## **Materials Advances**



### **PERSPECTIVE**

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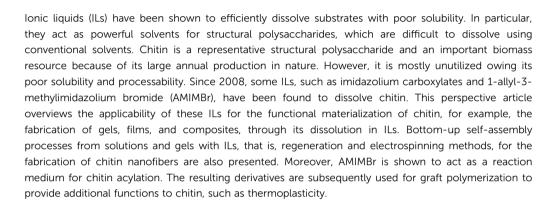
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### Application of ionic liquids for the functional materialization of chitin

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#### Introduction

Ionic liquids (ILs) are low melting point molten salts that form liquids at room temperature or even at temperatures below the boiling point of water. This specific property is attributed to the fact that the liquid state is thermodynamically favorable because of the large size (mainly the cations) and

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conformational flexibility of the ions, which results in small lattice enthalpies and large entropy changes that favor the liquid state (Fig. 1). ILs have been identified as powerful solvents for various organic and inorganic materials with poor solubilities. For example, ILs have been used to dissolve natural polysaccharides, which are often insoluble in water and common organic solvents.2

Natural polysaccharides are the most abundant organic substances and biomass resources on the earth and mainly serve two important roles: structural materials and suppliers of water and energy.<sup>3,4</sup> Cellulose and chitin are two representative structural polysaccharides with huge annual productions, which are mainly present in the cell walls of plants and the exoskeletons of crustaceans, respectively.<sup>5-8</sup> These polysaccharides are composed of repeating D-glucose and N-acetyl-D-glucosamine units, respectively, which are linked by  $\beta(1 \rightarrow 4)$ -glycosidic linkages (Fig. 2). These specific structures allow for the formation of numerous regularly controlled intra- and intermolecular hydrogen bonds, resulting in high crystallinity and extended fibrous

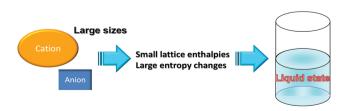


Fig. 1 Large ion sizes induce liquid state of ionic material.

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Fig. 2 Structures of cellulose and chitin

chain packing. Therefore, structural polysaccharides exhibit poor solubility in common solvents compared to that of energy storage polysaccharides, such as starch. Cellulose has traditionally been used in fibrous and structural applications, such as wood, paper, textiles, and furniture. Compared with cellulose, chitin is an unused biomass resource, mainly because of the presence of acetamido groups at the C-2 position in the N-acetyl-Dglucosamine units, which form strong intermolecular hydrogen bonds. Therefore, the solubility of chitin is inferior to that of cellulose, leading to poor processability and underutilization. However, chitin market was valued at \$42.29 million in 2020 and is projected to reach \$69.297 billion in 2028, growing at a CAGR (Compound Annual Growth Rate) of 5.07% from 2021 to 2028.9 The market demand is basically owing to the market growth of biopolymer processing, principally from its biological properties. 10 Rise in the chitin demand of uses as nanomaterials is also the key market driver.

Since an IL, 1-butyl-3-methylimidazolium chloride (BMIMCl), was reported to dissolve cellulose in 2002,11 numerous ILs have been used as solvents for cellulose and other polysaccharides. 12-21 Furthermore, ILs have been used as media for the derivatization, modification, and functionalization of cellulose. 14,17,18 Owing to its aforementioned poor solubility, ILs that can dissolve chitin were not found until 2008. 22-24 However, some ILs have been reported to dissolve chitin.<sup>25-28</sup> Moreover, these ILs are used to modify chitin, such as soft materialization, nanomaterialization, derivatization, and modification, to provide new functionalities on chitin. 26,29-34 This perspective article overviews recent progress in the efficient application of ILs for the functional materialization of chitin. Processability of chitin in different ILs into different materials and their potential applications are concisely summarized in the previously reported review articles. 26,32 Although deep eutectic solvents, analogous of ILs, have also been employed for the dissolution and subsequent materialization of chitin, 35-39 ILs have some advantages based on the viewpoint of a structural variety with wide range of properties.

### Dissolution of chitin in ILs

The dissolution of chitin in an IL was first reported using 1-butyl-3-methylimidazolium acetate (BMIMOAc, Fig. 3) in 2008. 40 Chitins of various origins and molecular weights were dissolved in BMIMOAc at 110 °C at satisfactory concentrations

 $R = -CH_2CH_2CH_2CH_3: 1-Butyl-3-methylimidazolium acetate (BMIMOAc) \\ R -CH_2CH=CH_2: 1-Allyl-3-methylimidazolium acetate (AMIMOAc) \\ R = -CH_2CH_3: 1-Ethyl-3-methylimidazolium acetate (EMIMOAc)$ 

1-Allyl-3-methylimidazolium bromide (AMIMBr)

Fig. 3 Representative ionic liquids (ILs), which dissolve chitin.

(maximum of 6 wt% chitin from crab shells). By cooling the chitin/BMIMOAc solutions to the ambient temperature, the corresponding gels were formed, which were further converted into chitin sponge and film materials by regeneration using a water or methanol coagulant. The dissolution behaviors of chitin in a series of alkylimidazolium chloride, dimethyl phosphate, and 1-allyl-3-methylimidazolium acetate (AMIMOAc, Fig. 3) ILs was explored. The former two series of ILs did not dissolve satisfactory amounts of chitin (less than 1.5 wt%), while the latter IL dissolved chitin to 5 wt%. 22 Ethyl-3-methylimidazolium acetate (EMIMOAc, Fig. 3) was also reported to dissolve chitin to certain concentrations, 41 and it was used to extract chitin from raw crustacean shells, such as shrimp shells. 1-Ethyl-3methylimidazolium alkanoates with different chain lengths and 1-ethyl-3-methylimidazolium/tetrabutylphosphonium amino acid salts were also found to dissolve chitin.42 The dissolution of chitin using cyclic ammonium acetates and lactates with different substituents was explored, which formed solutions containing 1-9 wt% chitin. 43 Chitin was also extracted from shrimp shells using several (hydroxy)ammonium acetate ILs. 44,45 Overall, the above dissolution studies of chitin in ILs suggest that ILs with carboxylate anions generally act as good solvents for chitin.

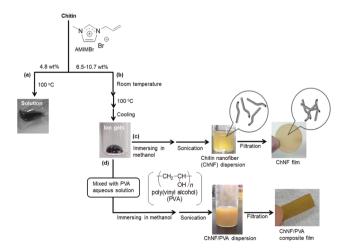


Fig. 4 (a) Dissolution and (b) gelation of chitin with AMIMBr and preparation of (c) self-assembled ChNF dispersion/film and (d) ChNF/PVA composite film (reprinted with permission from ref. 33. Copyright 2020 Elsevier; DOI: 10.1016/B978-0-12-817968-0.00002-0).

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Another type of IL, 1-allyl-3-methylimidazolium bromide (AMIMBr, Fig. 3), was used to dissolve chitin in concentrations of up to 4.8 wt% at 100  $^{\circ}$ C (Fig. 4a). <sup>46</sup> This IL can be prepared by quaternarization of 1-methylimidazole with ally bromide in good yield, which is the more facile procedure than that for the abovementioned imidazolium carboxylates. 47 The presence of a small quantity of bromide anions in 1-allyl-3-methylimidazolium chloride, which were generated by in situ anion exchange with 2-bromoethyl acetate as a bromide generator, enhanced the dissolution of chitin. 48 The facile extraction of chitin from crab shells was achieved using AMIMBr, followed by demineralization using citric acid. 49 Solubility of different chitin types in various ILs is comprehensively listed in the review article by Shamshina.<sup>27</sup>

Molecular dynamics (MD) simulations were conducted to evaluate the dissolution behavior of chitin crystals in AMIMBr, 48 and the results suggested that the chitin chains were peeled from the crystal surface, accompanied by cleavage of hydrogen bonds. The MD results also revealed the subsequent dissolution process, in which Br contributed to the cleavage of intermolecular hydrogen bonds, mostly by acetamido groups, whereas AMIM<sup>+</sup> prevented the return to the crystalline phase after peeling (Fig. 5). On the other hand, the MD results indicated that molecular chain peeling was not observed in BMIMCl. Although the weak hydrogen bonds by hydroxy groups were cleaved and division of a crystal into blocks occurred, the stacked chains were still maintained in BMIMCl, probably due to weaker capability of Cl<sup>-</sup> than Br<sup>-</sup> to cleave hydrogen bonds by acetamido groups.

When larger amounts of chitin (6.5-10.7 wt%) were successively immersed in AMIMBr at room temperature, heated at 100 °C, and cooled to room temperature, gel-like materials (ion gels) with high viscosities formed (Fig. 4b).46 The dynamic rheological measurements showed that the AMIMBr ILs containing 4.8 and 6.5 wt% chitin behaved as weak gels. ILs with hydroxide anions have been used to dissolve chitin by heating and subsequent deacetylation in solution. 50,51

## Soft materialization of chitin using ILs

Soft materials, such as gels and films, containing chitin have been fabricated through dissolution in ILs. As mentioned

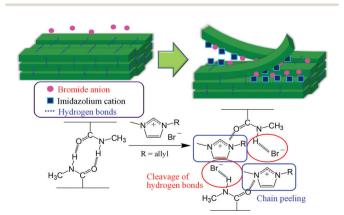


Fig. 5 Plausible dissolution process of chitin in AMIMBr predicted by molecular dynamics (MD) simulations.

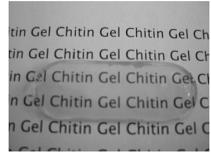


Fig. 6 Photograph of chitin gel prepared from solution in BMIMOAc (reprinted with permission from ref. 40. Copyright 2008 Elsevier; DOI: 10.1016/j.polymer.2008.03.027)

above, gels were obtained from chitin and BMIMOAc solutions (Fig. 6), which were further transformed into chitin sponges and films by regeneration using water and methanol coagulants, respectively. 40 Furthermore, the rheological properties of chitin ion gels containing BMIMOAc were investigated.52 The chain rigidity of chitin and the hydrogen bonds between chitin and BMIMOAc greatly affected the rheological behavior of the ion gels. EMIMOAc has also been used to produce gels, films, membranes, and fibers from chitin. 53-56 For example, chitin hydrogels were prepared through solution molding with EMI-MOAc, followed by water coagulation and washing.<sup>54</sup> ILs with alkanoate and amino acid anions induced the formation of chitin foils and coatings.42

Moreover, chitin-containing composite soft materials have been fabricated using ILs. For example, imidazolium acetates and alkanoates have been used to fabricate chitin/cellulose composite yarns, films, and hydrogels (Fig. 7).57-60 Chitin/ cellulose composite hydrogels prepared using EMIMOAc were investigated as novel electrolytes for electrochemical capacitors. 61 Chitin was also blended with poly(L-lactic acid) (PLLA) by co-dissolution in EMIMOAc, followed by extrusion into a water coagulation bath.62 Biocomposites containing chitin, PLLA, and hydroxyapatite were prepared by co-

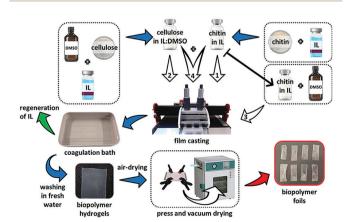


Fig. 7 Preparation procedure for chitin/cellulose composite film (reprinted from ref. 60 (Elsevier). This is an open access article distributed under the terms of the Creative Commons CC-BY license; DOI: 10.1016/ j.eurpolymj.2021.110681)

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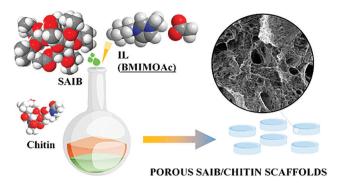
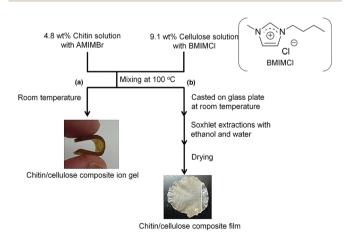


Fig. 8 Schematic representation for preparation of sucrose acetate isobutyrate (SAIB)-chitin scaffolds (reprinted with permission from ref. 65. Copyright 2020 American Chemical Society; DOI: 10.1021/acssuschemeng.0c00385).

dissolution in EMIMOAc, and their biocompatibility was evaluated. <sup>63</sup> The composites were appropriate for the growth and proliferation of osteocytes. Composite fibers composed of chitin and calcium alginate were fabricated by co-dissolution of chitin with alginic acid in EMIMOAc, followed by extrusion into a calcium chloride aqueous solution. <sup>64</sup> The resulting fibers were used as wound healing patches and exhibited excellent biocompatibility, fast reepithelization, and accelerated healing ability. BMIMOAc-mediated processing was investigated to fabricate sucrose acetate isobutyrate (SAIB)–chitin scaffolds (Fig. 8). <sup>65</sup> Human adipose stem cells were able to spread in the scaffolds, indicating the positive biological effect of SAIB. The result revealed that this type of scaffolds is practically applicable as a biomedical material.

A chitin/cellulose composite ion gel was fabricated from a solution containing two ILs, AMIMBr and BMIMCl (Fig. 9a). An AMIMBr solution containing 4.8 wt% chitin and a BMIMCl solution containing 9.1 wt% cellulose were first mixed at 100  $^{\circ}$ C to form a homogeneous solution. The chitin/cellulose composite ion gel was obtained by eliminating the excess ILs by



**Fig. 9** Preparation of (a) chitin/cellulose composite ion gel and (b) chitin/cellulose composite film using ionic liquids (reprinted with permission from ref. 33. Copyright 2020 Elsevier; DOI: 10.1016/B978-0-12-817968-0.00002-0).

incubating the solution at room temperature, followed by washing with ethanol. The composite ion gel was employed as a novel electrolyte for an electric double-layer capacitor after conversion into the corresponding acidic gel by treatment with a 2.0 mol L<sup>-1</sup> H<sub>2</sub>SO<sub>4</sub> aqueous solution. <sup>67-69</sup> The acidic composite gel electrolyte exhibited an excellent high-rate discharge capability over a wide range of current densities as well as in a H<sub>2</sub>SO<sub>4</sub> aqueous solution. In addition, the discharge capacitance of the test cell retained over 80% of its initial value after 105 cycles, even at a high current density of 5000 mA g<sup>-1</sup>. Chitin materials combined with ILs, which have excellent ionic conductivity, are thus suitable for electrochemical application. Casting the aforementioned homogeneous solution containing chitin/cellulose and AMIMBr/BMIMCl on a glass plate, followed by coagulation with ethanol and water, vielded a chitin/ cellulose composite film (Fig. 9b).66,70

# Chitin nanomaterials from IL solutions and gels

The preparation of nanoscale chitin assemblies, such as nanofibers and nanocrystals, is one of the most useful methods to practically transform chitin sources into functional materials because of the exceptional properties of bio-based nanomaterials, such as their lightweight, high tensile strength, low thermal expansion coefficient, biocompatibility, and nanosheet formability for sensing and electronic devices. The in addition to the conventional top-down approach, regenerative bottom-up approaches from solutions/gels with ILs have also been used to fabricate chitin nanofibers (ChNFs). Also, 24,33,84-86

ChNF dispersions were obtained by immersing the aforementioned 9.1-10.7 wt% chitin ion gels with AMIMBr in methanol at room temperature for 24 h to slowly regenerate chitin, followed by sonication (Fig. 4c).87 Nanofibers approximately 20-60 nm in width and several hundred nanometers in length were observed by scanning electron microscopy of the methanol dispersion, supporting the controlled self-assembly of ChNFs through regeneration from the ion gels. High entanglement of ChNFs occurred through the filtration of the dispersion to form a ChNF film. Self-assembled ChNFs with higher aspect ratios were obtained by regeneration using a CaBr<sub>2</sub>. 2H<sub>2</sub>O/methanol solution at an adjusted concentration. 88 Transmission electron microscopy observations of the self-assembled ChNFs identified bundle-like nanofiber morphologies, indicating that they were hierarchically constructed through the assembly of thinner fibrils.89 When the self-assembled ChNF film was treated with aqueous NaOH for partial deacetylation of acetamido groups and subsequently treated in a 1.0 mol  $L^{-1}$ aqueous acetic acid with ultrasonication, the bundles disintegrated through cationization and electrostatic repulsion of the generated amino groups, yielding a dispersion of thinner nanofibers, named "scaled-down (SD)-ChNFs."90 The SD-ChNFs were isolated through filtration of the dispersion to fabricate a highly flexible film that bent and twisted easily.

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Nano-scaled materials, such as ChNFs, are good candidates to be used in composites both physically and chemically combined with other polymeric components. For example, the self-assembled ChNFs were composited with poly(vinyl alcohol) (PVA) to fabricate composite films (Fig. 4d). 87 Solutions comprising the desired amounts of PVA in hot water were first added to the chitin ion gel with AMIMBr. The same regeneration and filtration procedures as described above were used to produce ChNF/PVA composite films. Tensile testing of the composite films indicated the mechanical properties improved with increasing ratio of PVA to chitin.

Self-assembled ChNFs have been employed as stabilizers for the Pickering emulsion polymerization of styrene, 91 in which Pickering emulsions of any type (oil-in-water, water-in-oil, or even multicomponent emulsions) are stabilized by solid particles or other types of solid materials in place of the surfactants in general emulsions. Prior to emulsion polymerization, anionic carboxylate groups (maleyloyl groups) were introduced on ChNFs by reaction with maleic anhydride in the presence of perchloric acid to enhance dispersibility in aqueous ammonia. Radical polymerization using potassium persulfate as an initiator was then performed at 70 °C in the emulsion, in which styrene droplets were stably surrounded by ChNFs in aqueous ammonia, to fabricate composite particles. When the Pickering emulsion polymerization of styrene was conducted using the aforementioned SD-ChNFs as stabilizers, the composite particles were smaller than those produced using the self-assembled ChNFs.92 The composite particles were easily converted into hollow ChNF-based particles by solubilizing the styrene core with toluene. In addition to the introduction of anionic maleyloyl groups on the ChNFs, polymerizable methacryloyl groups were substituted on the ChNFs as a secondary functionalization, which could be copolymerized with styrene to stabilize the hollow structure. Hollow particles were fabricated by using toluene to solubilize the polystyrene inside the composite particles fabricated using the bifunctional ChNF stabilizers, which stably redispersed in water.93

Self-assembled ChNFs have been employed as reinforcements through combination with other polymers. 86 Because chitin is regarded as a cationic polymer owing to the presence of several percent of free amino groups in the total repeating units due to deacetylation of acetamido groups, self-assembled ChNFs were composited with an anionic polysaccharide, carboxymethyl cellulose (CMC), through electrostatic interactions. 94 Cast CMC films were immersed in self-assembled ChNF/methanol dispersions with different contents, followed by centrifugation and drying, to fabricate the ChNF-reinforced CMC films. The quantity of ChNFs in the composite film strongly affected the enhancement of the mechanical properties under tension, supporting their reinforcement effect. The reinforcement effect of the self-assembled ChNFs was also observed by adding them to hydrogels of another anionic polysaccharide, xanthan gum. 95 A composite film of the aforementioned SD-ChNFs with 1-carrageenan, a sulfated anionic polysaccharide, was also produced via electrostatic interactions.90

Self-assembled ChNF-reinforced cellulose and natural rubber (NR) films/sheets have also been fabricated. 96,97 When cellulose/

BMIMCl ion gels, which easily formed from solution, 98 were immersed in self-assembled ChNF/methanol dispersions, the two polysaccharides were composited through the regeneration of cellulose to fabricate ChNF-reinforced cellulose films. Selfassembled ChNF dispersions with aqueous ammonia were mixed with NR latex stabilized with aqueous ammonia, followed by drying under reduced pressure, to obtain ChNF-reinforced NR sheets. Tensile testing of the ChNF/cellulose films and ChNF/NR sheets demonstrated the reinforcement effect of ChNFs. The above ChNF/NR dispersions in aqueous ammonia were heated to evaporate the ammonia, followed by lyophilization, resulting in a porous material.<sup>97</sup> By evaporating the ammonia stabilizer, ChNFs aggregated with the NR and then agglomerated to form spaces between them, resulting in the porous morphology.

Self-assembled ChNF-graft-synthetic polymer composite films were fabricated by surface-initiated graft polymerization (Fig. 10).84 ChNF-graft-biodegradable polyester and polypeptide composite films were prepared by surface-initiated ring-opening graft polymerization of the corresponding cyclic monomers, that is, L-lactide (LA)/ε-caprolactone (CL) and γ-benzyl L-glutamate-NCA (BLG-NCA), initiated from hydroxy and amino groups, respectively, present on the self-assembled ChNF films. 99,100 Surface-initiated graft atom transfer radical polymerization (ATRP) from ChNF macroinitiators has also been conducted. ATRP is a versatile living radical polymerization technique, which has been employed to synthesize a wide range of welldefined polymeric materials. 101,102 Because ATRP is initiated by α-haloalkylacyl groups, the ChNF macroinitiator film with the initiating groups was prepared by the reaction of hydroxy groups on the self-assembled ChNFs with  $\alpha$ -bromoisobutyryl bromide. The surface-initiated ATRP of 2-hydroxyethyl acrylate (HEA) from the resulting macroinitiator film was then demonstrated in the presence of CuBr (catalyst)/2,2'-bipyridine (ligand) in 3 wt% LiCl/ N,N-dimethylacetamide at 60 °C to fabricate ChNF-graft-polyHEA films. 103 Tensile testing of the fabricated films indicated larger elongation values at break compared to that of the original ChNF film, which increased with increasing graft chain length. Surface-initiated graft ATRP of methyl methacrylate (MMA) on the ChNF macroinitiator under dispersion conditions was also

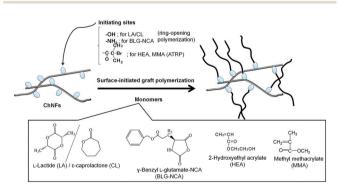


Fig. 10 Surface-initiated graft ring opening (co)polymerization and atom transfer radical (ATRP) polymerization from appropriate initiating sites on self-assembled chitin nanofibers (ChNF) (reprinted with permission from Copyright 2020 Elsevier; DOI: 10.1016/B978-0-12-817968-0.00002-0).

Perspective

Weaker

Weaker

Chilin Fibers

E 60

Stronger
Chilin-PLA
Composite Fibers

Fig. 11 Image for fabrication of ChNF-poly(lactic acid) (PLA) nanocomposite fibers and stress–strain curves under tensile mode (reprinted with permission from ref. 62. Copyright 2018 American Chemical Society; DOI: 10.1021/acssuschemeng.8b01554).

attempted, followed by entanglement of the products by filtration, to fabricate ChNF-graft-polyMMA films. 104

Chitin solutions in EMIMOAc have been electrospun to fabricate ChNFs *via* a bottom-up process. 105-107 Because EMIMOAc does not evaporate, a coagulation bath is required to remove EMIMOAc and precipitate ChNFs. The high viscosities of the solutions allow electrospinning of solutions with low chitin concentrations (less than 1 wt%) to produce thin fibers without adjusting the electrospinning parameters. This process has been extended to the electrospinning of chitin composites with other polymers, such as lignin, cellulose, and PLLA (Fig. 11). 1162,108 These results strongly suggest that besides regeneration as above mentioned, electrospinning is also a powerful technique for controlled nano-assembly from chitin according to bottom-up approach.

## Derivatization and modification of chitin in ILs

Derivatization of polysaccharides, such as acylation (ester derivatization), has been identified as a useful functional materialization, which provides high-performance properties and is

Fig. 12 (a) Acylation of chitin in AMIMBr and (b) substituents for mixed chitin esters.

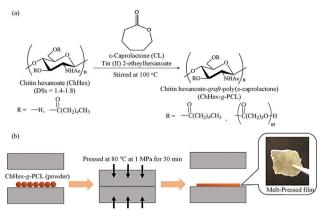
used in practical applications such as bio-based thermoplastics, as represented by cellulose acylates. 109-111 Chitin acylates have not been employed in practical applications because efficient and practical acylation methods for chitin from native sources have not been sufficiently developed. 6,112-115 AMIMBr has been found to act as a good reaction medium for the acylation of chitin. 116 For example, the acetylation of chitin in AMIMBr was investigated to obtain chitin acetate. 117 Reactions using acetic anhydride (20 equiv. with a repeating unit) in a 2 wt% AMIMBr solution at 60-100 °C for 24 h yielded degrees of derivatization (DSs) of 1.82 to 1.90 (Fig. 12a, m = 0), indicating the occurrence of almost quantitative acetylation (the highest DS value for chitin is 2). In contrast, ILs with carboxylate anions, such as acetate and alkanoates, are not considered as suitable media for acylation because of their higher basicity compared to that of bromide in AMIMBr, resulting in potential reactivity with acylation reagents. Accordingly, acetylation of chitin using 20 equiv. of acetic anhydride in BMIMOAc at 100 °C for 24 h vielded a lower DS value (0.81) than that obtained with AMIMBr (1.90) under the same conditions.

When the synthesis of various chitin acylates with different substituents was attempted in AMIMBr, the following optimal conditions were found. Using acyl chlorides in the presence of pyridine as a base and *N*,*N*-dimethyl-4-aminopyridine (DMAP) as a catalyst at 100 °C for 24 h produced derivatives with quantitative DS values (Fig. 12a). To confirm the efficient acylation of chitin using acyl chlorides in AMIMBr, a mixture of hexanoyl chloride and AMIMBr was heated at 100 °C in the absence of chitin. H NMR analysis supported the occurrence of halogen exchange between hexanoyl chloride and AMIMBr in the mixture to produce hexanoyl bromide *in situ* (Fig. 13). Accordingly, acyl bromides efficiently react with hydroxy groups in chitin owing to their higher reactivity compared to that of acyl chlorides to obtain high-DS products.

The above chitin acylates (single chitin esters) did not exhibit specific processability, such as film formability. The synthesis of mixed chitin esters with two different substituents was then conducted using the same acylation manner in AMIMBr to provide derivatives with specific processability. This attempt was based on the fact that mixed cellulose esters exhibit a lower viscous flow temperature compared to that of single cellulose esters, leading to superior melt processability and thermoplasticity. A stearoyl group was selected as the first substituent, which was combined with different long fatty and bulky acyl groups as the second substituent (Fig. 12b). The almost quantitative DS values achieved by stearoyl with oleoyl or some bulky acyl groups resulted from acylation using the corresponding acyl chlorides in the presence of pyridine/DMAP

**Fig. 13** Halogen exchange reaction between hexanoyl chloride and AMIMBr to produce hexanoyl bromide (reprinted with permission from ref. 119. Copyright 2021 Elsevier; DOI: 10.1016/j.ijbiomac.2021.09.044).

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(a) Surface-initiated ring-opening graft polymerization of ε-caprolactone (CL) from chitin hexanoates (ChHexs) to obtain chitin hexanoate-graft-poly(ε-caprolactone) (ChHex-g-PCL), and (b) melt-pressing experiment of ChHex-g-PCL (reprinted with permission from ref. 121. Copyright 2022 Elsevier; DOI: 10.1016/j.carbpol.2021.119024).

in AMIMBr. All products exhibited film formability by casting from solutions in chloroform or chloroform/trifluoroacetic acid solvents, but they did not exhibit thermoplasticity. There is a possibility for exhibiting thermostasticity from mixed chitin esters with different substituents and their DSs by additional investigations in the future.

To achieve thermoplasticity, surface-initiated ring-opening graft polymerizations of CL from the hydroxy groups of chitin hexanoates with DS values in the 1.4-1.8 range, which were prepared by the acylation of chitin with hexanoyl chloride in AMIMBr, were performed (Fig. 14a). 121 Surface-initiated ring-opening graft polymerization of CL was conducted in the presence of tin(II) 2-ethylhexanoate as a catalyst at 100 °C to produce chitin hexanoate-graft-poly(ε-caprolactone) (ChHex-g-PCL). The reaction conditions strongly affected the molar substitution and degree of polymerization of the PCL graft chains. Longer PCL graft chains formed crystalline structures, and the ChHex-g-PCL materials largely contained uncrystallized chitin chains. Accordingly, these ChHex-g-PCL materials exhibited melting points associated with the PCL graft chain length, allowing the formation of films by melt pressing at 80 °C and 1 MPa for 30 min (Fig. 14b). This study provides the first example of thermoplastic chitin derivatives.

The above acylation method for chitin in AMIMBr was extended to synthesize a chitin macroinitiator for subsequent graft ATRP. 122 The chitin macroinitiator modified with initiating groups was synthesized by the acylation of hydroxy groups in chitin with 2-bromopropyl bromide in AMIMBr according to the above reaction mechanism (Fig. 15a). The DS value of the product, which was synthesized by a reaction using 30 equiv. of 2-bromopropyl bromide with a repeating unit at 100 °C for 24 h, was estimated to be 1.86 by <sup>1</sup>H NMR analysis. The graft polymerization of styrene was then conducted by ATRP from the resulting chitin macroinitiator using CuBr and N,N,N'N',N",N"pentamethyldiethylene triamine (PMDETA) as the catalyst system to obtain chitin-graft-polystyrene (Fig. 15b). The grafted polystyrene chains were separated from the products by alkaline

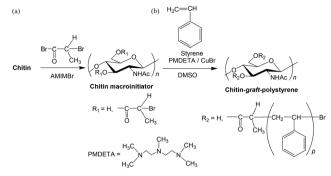


Fig. 15 (a) Synthesis of chitin macroinitiator in AMIMBr and (b) following graft atom transfer radical polymerization (ATRP) of styrene (reprinted with permission from ref. 116. Copyright 2019 Elsevier; DOI: 10.1016/ i.ijbiomac.2018.11.165).

hydrolysis of ester linkages at the root of the chitin chain to estimate their  $M_n$  values using size exclusion chromatography (SEC). The SEC results suggested the  $M_{\rm n}$  value increased with increasing monomer/initiating site feed ratio, indicating graft ATRP progressed in a living manner.

### Conclusions

This perspective article presents the efficient functional materialization of chitin through dissolution in ILs. Because of the poor solubility and processability of chitin, ILs have attracted increasing attention for the development of new uses and applications of chitin. Unique forms, such as gels and films, can be fabricated from chitin solutions in ILs. The regeneration and electrospinning of solutions/gels provide a bottom-up selfassembly approach to easily fabricate ChNFs. The resulting chitin materials have been used to prepare composite materials. Acylation of chitin efficiently occurs in AMIMBr to produce various chitin acylates. This article illustrates that the possible applications of chitin as a material have been widely extended through the use of ILs, which are now identified as an important research field regarding chitin. Indeed, the dissolution of chitin in ILs has provided a certain number of materials with tailored and improved properties, enhancing the potential use of chitin in different applications, such as in biomedical, tissue engineering, and green/sustainable fields as some previous review articles have stated. 26,28,32-34,86 Based on environmentally aspect, harmful chemicals, currently used in industrial processes, should be replaced by solvents that show suitability for the green chemistry approach, such as ILs, and that the use of greener ILs in applications of chitin in functional material field will gradually increase. Overall, further research will be conducted, including the development of new greener ILs for chitin solvents and additional approaches to add new functions, which can be employed in practical materials and can lead to their commercialization in the future.

#### Author contributions

The author designed and wrote the manuscript.

### Conflicts of interest

The author declares no conflict of interest.

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### References

- 1 J. G. Yao, P. S. Fennell and J. P. Hallett, RSC Energy and Environment Series, 2020, vol. 2020, pp. 69-105.
- 2 X. Yang, C. Qiao, Y. Li and T. Li, React. Funct. Polym., 2016, **100**, 181-190.
- 3 C. Schuerch, in Encyclopedia of Polymer Science and Engineering, ed. H. F. Mark, N. Bilkales and C. G. Overberger, John Wiley & Sons, New York, 2nd edn, 1986, vol. 13, pp. 87-162.
- 4 E. H. Song, J. Shang and D. M. Ratner, in Polymer Science: A Comprehensive Reference, ed. K. Matyjaszewski and M. Möller, Elsevier, Amsterdam, 2012, pp. 137-155.
- 5 D. Klemm, B. Heublein, H. P. Fink and A. Bohn, Angew. Chem., Int. Ed., 2005, 44, 3358-3393.
- 6 K. Kurita, Mar. Biotechnol., 2006, 8, 203-226.
- 7 M. Rinaudo, Prog. Polym. Sci., 2006, 31, 603-632.
- 8 C. K. S. Pillai, W. Paul and C. P. Sharma, *Prog. Polym. Sci.*, 2009, 34, 641-678.
- 9 Global Chitin Market Size By Derivative Type, By End-User, By Geographic Scope And Forecast, Verified Market Research, Lewws, DE, USA, 2021.
- 10 R. A. A. Muzzarelli, M. El Mehtedi and M. Mattioli-Belmonte, Mar. Drugs, 2014, 12, 5468-5502.
- 11 R. P. Swatloski, S. K. Spear, J. D. Holbrey and R. D. Rogers, J. Am. Chem. Soc., 2002, 124, 4974-4975.
- 12 O. A. El Seoud, A. Koschella, L. C. Fidale, S. Dorn and T. Heinze, Biomacromolecules, 2007, 8, 2629-2647.
- 13 T. Liebert and T. Heinze, BioResources, 2008, 3, 576-601.
- 14 L. Feng and Z. I. Chen, J. Mol. Liq., 2008, 142, 1-5.
- 15 A. Pinkert, K. N. Marsh, S. S. Pang and M. P. Staiger, Chem. Rev., 2009, 109, 6712-6728.
- 16 M. E. Zakrzewska, E. Bogel-Łukasik and R. Bogel-Łukasik, Energy Fuels, 2010, 24, 737-745.
- 17 M. Gericke, P. Fardim and T. Heinze, *Molecules*, 2012, 17, 7458-7502.
- 18 M. Isik, H. Sardon and D. Mecerreyes, Int. J. Mol. Sci., 2014, 15, 11922-11940.
- 19 J. Zhang, J. Wu, J. Yu, X. Zhang, J. He and J. Zhang, Mater. Chem. Front., 2017, 1, 1273-1290.
- 20 F. Hermanutz, Μ. P. Vocht, N. Panzier M. R. Buchmeiser, Macromol. Mater. Eng., 2019, 304, 1800450.

- 21 C. Verma, A. Mishra, S. Chauhan, P. Verma, V. Srivastava, M. A. Quraishi and E. E. Ebenso, Sustainable Chem. Pharm., 2019, 13, 100162.
- 22 W. T. Wang, J. Zhu, X. L. Wang, Y. Huang and Y. Z. Wang, J. Macromol. Sci. Phys., 2010, 49, 528-541.
- 23 J. Kadokawa, Green Sustainable Chem., 2013, 03, 19-25.
- 24 J. Kadokawa, RSC Adv., 2015, 5, 12736-12746.
- 25 M. M. Jaworska, T. Kozlecki and A. Gorak, J. Polym. Eng., 2012, 32, 67-69.
- 26 S. S. Silva, J. F. Mano and R. L. Reis, Green Chem., 2017, 19, 1208-1220.
- 27 J. L. Shamshina, Green Chem., 2019, 21, 3974-3993.
- 28 S. K. Singh, Int. J. Biol. Macromol., 2019, 132, 265-277.
- 29 R. A. A. Muzzarelli, Mar. Drugs, 2011, 9, 1510-1533.
- 30 J. Kadokawa, in RSC Smart Materials, ed. A. Eftekhari, Royal Society of Chemistry, London, 2018, pp. 319-341.
- 31 J. L. Shamshina, O. Zavgorodnya and R. D. Rogers, Advances in. Biochemical Engineering/Biotechnology, 2019, vol. 168, pp. 177-198.
- 32 J. L. Shamshina, P. Berton and R. D. Rogers, ACS Sustainable Chem. Eng., 2019, 7, 6444-6457.
- 33 J. Kadokawa, in Handbook of Chitin and Chitosan, ed. S. Gopi, S. Thomas and A. Pius, Elsevier, 2020, pp. 47-60.
- 34 J. L. Shamshina, A. Kelly, T. Oldham and R. D. Rogers, Ebviron. Chem. Lett., 2020, 18, 53-60.
- 35 M. Sharma, C. Mukesh, D. Mondal and K. Prasad, RSC Adv., 2013, 3, 18149-18155.
- 36 C. Mukesh, D. Mondal, M. Sharma and K. Prasad, Carbohydr. Polym., 2014, 103, 466-471.
- 37 S. Idenoue, K. Yamamoto and J. Kadokawa, ChemEngineering, 2019, 3, 90, DOI: 10.3390/chemengineering3040090.
- 38 J. Kadokawa, S. Idenoue and K. Yamamoto, ACS Sustainable Chem. Eng., 2020, 8, 8402-8408.
- 39 N. Özel and M. Elibol, Carbohydr. Polym., 2021, 262, 117942.
- 40 Y. Wu, T. Sasaki, S. Irie and K. Sakurai, Polymer, 2008, 49, 2321-2327.
- 41 Y. Qin, X. M. Lu, N. Sun and R. D. Rogers, Green Chem., 2010, 12, 968-971.
- 42 P. Walther, A. Ota, A. Müller, F. Hermanutz, F. Gähr and M. R. Buchmeiser, Macromol. Mater. Eng., 2016, 301, 1337-1344.
- 43 M. M. Jaworska, I. Stępniak, M. Galiński, D. Kasprzak, D. Binias and A. Górak, Carbohydr. Polym., 2018, 202, 397-403.
- 44 L. D. Tolesa, B. S. Gupta and M.-J. Lee, Int. J. Biol. Macromol., 2019, 130, 818-826.
- 45 J. L. Shamshina, P. S. Barber, G. Gurau, C. S. Griggs and R. D. Rogers, ACS Sustainable Chem. Eng., 2016, 4, 6072-6081.
- 46 K. Prasad, M. Murakami, Y. Kaneko, A. Takada, Y. Nakamura and J. Kadokawa, Int. J. Biol. Macromol., 2009, 45, 221-225.
- 47 H. Zhang, Y. Xu, Y. Li, Z. Lu, S. Cao, M. Fan, L. Huang and L. Chen, Polymers, 2017, 9, 526.
- 48 T. Uto, S. Idenoue, K. Yamamoto and J. Kadokawa, Phys. Chem. Chem. Phys., 2018, 20, 20669-20677.
- 49 T. Setoguchi, T. Kato, K. Yamamoto and J. I. Kadokawa, Int. J. Biol. Macromol., 2012, 50, 861-864.

50 M. Shimo, M. Abe and H. Ohno, ACS Sustainable Chem. Eng., 2016, 4, 3722-3727.

**Materials Advances** 

- 51 Q. Ma, X. Gao, X. Bi, Q. Han, L. Tu, Y. Yang, Y. Shen and M. Wang, Carbohydr. Polym., 2020, 230, 115605.
- 52 L. Deng and L.-M. Zhang, Colloids Surf., A, 2020, 586, 124220.
- 53 D. G. Ramírez-Wong, M. Ramírez-Cardona, R. J. Sánchez-Leija, A. Rugerio, R. A. Mauricio-Sánchez, M. A. Hernández-Landaverde, A. Carranza, J. A. Pojman, A. M. Garay-Tapia, E. Prokhorov, J. D. Mota-Morales and G. Luna-Bárcenas, Green Chem., 2016, 18, 4303-4311.
- 54 X. Shen, J. L. Shamshina, P. Berton, J. Bandomir, H. Wang, G. Gurau and R. D. Rogers, ACS Sustainable Chem. Eng., 2016, 4, 471-480.
- 55 C. King, J. L. Shamshina, G. Gurau, P. Berton, N. F. A. F. Khan and R. D. Rogers, Green Chem., 2017, 19, 117-126.
- 56 J. Chakravarty, M. F. Rabbi, N. Bach, V. Chalivendra, C.-L. Yang and C. J. Brigham, Carbohydr. Polym., 2018, **198**, 443-451.
- 57 K. Mundsinger, A. Müller, R. Beyer, F. Hermanutz and M. R. Buchmeiser, Carbohydr. Polym., 2015, 131, 34-40.
- 58 Y. Duan, A. Freyburger, W. Kunz and C. Zollfrank, Carbohydr. Polym., 2018, 192, 159-165.
- 59 P. Berton, X. Shen, R. D. Rogers and J. L. Shamshina, Ind. Eng. Chem. Res., 2019, 58, 19862-19876.
- 60 D. Kasprzak and M. Galiński, Eur. Polym. J., 2021, **158**, 110681.
- 61 D. Kasprzak and M. Galiński, J. Solid State Electrochem., 2021, 25, 2549-2563.
- 62 J. L. Shamshina, O. Zavgorodnya, P. Berton, P. K. Chhotaray, H. Choudhary and R. D. Rogers, ACS Sustainable Chem. Eng., 2018, 6, 10241-10251.
- 63 J. Chakravarty, M. F. Rabbi, V. Chalivendra, T. Ferreira and C. J. Brigham, Int. J. Biol. Macromol., 2020, 151, 1213-1223.
- 64 J. L. Shamshina, G. Gurau, L. E. Block, L. K. Hansen, C. Dingee, A. Walters and R. D. Rogers, J. Mater. Chem. B, 2014, 2, 3924-3936.
- 65 C. Gonçalves, S. S. Silva, J. M. Gomes, I. M. Oliveira, R. F. Canadas, F. R. Maia, H. Radhouani, R. L. Reis and J. M. Oliveira, ACS Sustainable Chem. Eng., 2020, 8, 3986-3994.
- 66 A. Takegawa, M. Murakami, Y. Kaneko and J. Kadokawa, Carbohydr. Polym., 2010, 79, 85-90.
- 67 S. Yamazaki, A. Takegawa, Y. Kaneko, J. Kadokawa, M. Yamagata and M. Ishikawa, Electrochem. Commun., 2009, 11, 68-70.
- 68 S. Yamazaki, A. Takegawa, Y. Kaneko, J. Kadokawa, M. Yamagata and M. Ishikawa, J. Power Sources, 2010, 195, 6245-6249.
- 69 S. Yamazaki, A. Takegawa, Y. Kaneko, J. Kadokawa, M. Yamagata and M. Ishikawa, J. Electrochem. Soc., 2010, 157, A203-A208.
- 70 J. Kadokawa, K. Hirohama, S. Mine, T. Kato and K. Yamamoto, J. Polym. Environ., 2012, 20, 37-42.
- 71 J. You, M. Li, B. Ding, X. Wu and C. Li, *Adv. Mater.*, 2017, 29, 1606895.

- 72 M. Anraku, R. Tabuchi, S. Ifuku, T. Nagae, D. Iohara, H. Tomida, K. Uekama, T. Maruyama, S. Miyamura, F. Hirayama and M. Otagiri, Carbohydr. Polym., 2017, 161, 21-25.
- 73 R. Koizumi, K. Azuma, H. Izawa, M. Morimoto, K. Ochi, T. Tsuka, T. Imagawa, T. Osaki, N. Ito, Y. Okamoto, H. Saimoto and S. Ifuku, Int. J. Mol. Sci., 2017, 18, 11.
- 74 C. C. Satam, C. W. Irvin, A. W. Lang, J. C. R. Jallorina, M. L. Shofner, J. R. Reynolds and J. C. Meredith, ACS Sustainable Chem. Eng., 2018, 6, 10637-10644.
- 75 N. E. Mushi, T. Nishino, L. A. Berglund and Q. Zhou, ACS Sustainable Chem. Eng., 2019, 7, 1692-1697.
- 76 T. Naghdi, H. Golmohammadi, H. Yousefi, M. Hosseinifard, U. Kostiv, D. Horák and A. Merkoçi, ACS Appl. Mater. Interfaces, 2020, 12, 15538-15552.
- 77 T. Yang, H. Qi, P. Liu and K. Zhang, ChemPlusChem, 2020, 85, 1081-1088.
- 78 S. Ifuku and H. Saimoto, Nanoscale, 2012, 4, 3308-3318.
- 79 S. Ifuku, Kobunshi Ronbunshu, 2012, **69**, 460-467.
- 80 S. Ifuku, Molecules, 2014, 19, 18367-18380.
- 81 Y. Fan, T. Saito and A. Isogai, Biomacromolecules, 2008, 9, 192-198.
- 82 Y. M. Fan, T. Saito and A. Isogai, Carbohydr. Polym., 2009, 77, 832-838.
- 83 Y. M. Fan, H. Fukuzumi, T. Saito and A. Isogai, Int. J. Biol. Macromol., 2012, 50, 69-76.
- 84 J. Kadokawa, Coatings, 2016, 6, 27.
- 85 J. Kadokawa, Pure Appl. Chem., 2016, 88, 621-629.
- 86 J. Kadokawa, Polymers, 2021, 13, 3548.
- 87 J. Kadokawa, A. Takegawa, S. Mine and K. Prasad, Carbohydr. Polym., 2011, 84, 1408-1412.
- 88 R. Tajiri, T. Setoguchi, S. Wakizono, K. Yamamoto and J. Kadokawa, J. Biobased Mater. Bioenergy, 2013, 7, 655-659.
- 89 J. Kadokawa, A. Kawano and K. Yamamoto, ChemistrySelect, 2019, 4, 797-801.
- 90 T. Hashiguchi, K. Yamamoto and J. Kadokawa, Carbohydr. Polym., 2021, 270, 118369.
- 91 S. Noguchi, K. Sato, K. Yamamoto and J. Kadokawa, Int. J. Biol. Macromol., 2019, 126, 187-192.
- 92 R. Watanabe, K. Izaki, K. Yamamoto and J. Kadokawa, Coatings, 2021, 11, 672.
- 93 S. Noguchi, K. Yamamoto and J. Kadokawa, Int. J. Biol. Macromol., 2019, 157, 680-686.
- 94 D. Hatanaka, K. Yamamoto and J. Kadokawa, Int. J. Biol. Macromol., 2014, 69, 35-38.
- 95 J. Kadokawa, S. Noguchi, T. Gotanda, A. Kawano and K. Yamamoto, Polym. Bull., 2020, 77, 4095-4103.
- 96 J. Kadokawa, R. Endo, D. Hatanaka and K. Yamamoto, J. Polym. Environ., 2015, 23, 348-355.
- 97 A. Kawano, K. Yamamoto and J. Kadokawa, Biomolecules, 2017, 7, 47.
- 98 J. Kadokawa, M. Murakami and Y. Kaneko, Carbohydr. Res., 2008, 343, 769-772.
- 99 T. Setoguchi, K. Yamamoto and J. Kadokawa, Polymer, 2012, 53, 4977-4982.
- 100 J. Kadokawa, T. Setoguchi and K. Yamamoto, Polym. Bull., 2013, 70, 3279-3289.

Perspective

101 M. Kamigaito, T. Ando and M. Sawamoto, *Chem. Rev.*, 2001, **101**, 3689–3745.

- 102 K. Matyjaszewski, Macromolecules, 2012, 45, 4015-4039.
- 103 K. Yamamoto, S. Yoshida and J. Kadokawa, *Carbohydr. Polym.*, 2014, **112**, 119–124.
- 104 R. Endo, K. Yamamoto and J. Kadokawa, *Fibers*, 2015, 3, 338–347.
- 105 P. S. Barber, C. S. Griggs, J. R. Bonner and R. D. Rogers, Green Chem., 2013, 15, 601–607.
- 106 J. L. Shamshina, O. Zavgorodnya, J. R. Bonner, G. Gurau, T. Di Nardo and R. D. Rogers, *ChemSusChem*, 2017, **10**, 106–111.
- 107 O. Zavgorodnya, J. L. Shamshina, J. R. Bonner and R. D. Rogers, *ACS Sustainable Chem. Eng.*, 2017, 5, 5512–5519.
- 108 J. L. Shamshina, O. Zavgorodnya, H. Choudhary, B. Frye, N. Newbury and R. D. Rogers, ACS Sustainable Chem. Eng., 2018, 6, 14713–14722.
- 109 Q. Tang and G. Huang, Mini-Rev. Med. Chem., 2016, 16, 1244–1257.
- 110 K. J. Edgar, C. M. Buchanan, J. S. Debenham, P. A. Rundquist, B. D. Seiler, M. C. Shelton and D. Tindall, *Prog. Polym. Sci.*, 2001, 26, 1605–1688.

- 111 M. Kostag, M. Gericke, T. Heinze and O. A. El Seoud, *Cellulose*, 2019, **26**, 139–184.
- 112 M. N. V. Ravi Kumar, React. Funct. Polym., 2000, 46, 1-27.
- 113 H. Sashiwa and S.-I. Aiba, *Prog. Polym. Sci.*, 2004, **29**, 887–908.
- 114 K. Kurita, Prog. Polym. Sci., 2001, 26, 1921-1971.
- 115 M. Morimoto, H. Saimoto and Y. Shigemasa, *Trends Glycosci. Glycotechnol.*, 2002, **14**, 205–222.
- 116 J. Kadokawa, Int. J. Biol. Macromol., 2019, 123, 732-737.
- 117 S. Mine, H. Izawa, Y. Kaneko and J. Kadokawa, *Carbohydr. Res.*, 2009, 344, 2263–2265.
- 118 H. Hirayama, J. Yoshida, K. Yamamoto and J. Kadokawa, *Carbohydr. Polym.*, 2018, **200**, 567–571.
- 119 K. Kohori, H. Hirayama, K. Yamamoto and J. Kadokawa, *Int. J. Biol. Macromol.*, 2021, **190**, 763–768.
- 120 X. Wang, Y. Wang, Y. Xia, S. Huang, Y. Wang and Y. Qiu, *Text. Res. J.*, 2018, **88**, 1491–1504.
- 121 A. Nakashima, K. Kohori, K. Yamamoto and J. Kadokawa, *Carbohydr. Polym.*, 2022, **280**, 119024.
- 122 K. Yamamoto, S. Yoshida, S. Mine and J. Kadokawa, *Polym. Chem.*, 2013, 4, 3384–3389.