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# A facile acid treatment for P25 modification with enhanced photocatalytic H<sub>2</sub> evolution – effect of Brønsted acid sites and oxygen vacancies†

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Acid treated P25 catalysts (HP25-x, x represents the treatment temperature) were successfully fabricated via a simple soaking and drying process for photocatalytic H<sub>2</sub> production. After acid treatment, a substantial number of –OH groups were obtained on the surface of HP25-60, which can act as Brønsted acid sites providing hydrogen protons for H<sub>2</sub> generation. Besides, more oxygen vacancies and Ti<sup>3+</sup> sites were also created on the HP25-60 surface after acid treatment, which could enhance the carrier separation and transfer efficiency by capturing more electrons and holes, respectively. The H<sub>2</sub> production over HP25-60 reached 4853  $\mu$ mol h<sup>-1</sup> with Pt as a co-catalyst, which is 7.1 times higher than 679  $\mu$ mol h<sup>-1</sup> of bare TiO<sub>2</sub>. Three favorable factors for H<sub>2</sub> photogeneration, *i.e.* Brønsted acid sites, oxygen vacancies and Ti<sup>3+</sup> sites, can be introduced on a catalyst surface via a simple one-step acid strategy, which can be applied as a universally-applicable modification method for valence-alternative metal oxide semiconductors for enhanced H<sub>2</sub> photogeneration.

# 1. Introduction

Motivated by the increasing demand for a renewable and clean energy source in the future, hydrogen production from water-splitting by solar energy has been the subject of extensive investigations. Femiconductor photocatalysts are the key part for H<sub>2</sub> photogeneration, A,8 and among multifarious semiconductors, TiO<sub>2</sub> has been proven to be the most suitable candidate for H<sub>2</sub> photogeneration as a result of its high chemical stability, low cost, and strong oxidizing and reducing ability. Unfortunately, the photocatalytic H<sub>2</sub>-production efficiency on bare TiO<sub>2</sub> is very low due to the rapid carrier recombination and the large H<sub>2</sub> production overpotential. To address these limitations, a number of efforts have been made, including noble-metal deposition, semiconductor composition, dye sensitization, io ion doping, setc.

Besides the modification strategies mentioned above, acid treatment of TiO<sub>2</sub> is another simple and effective modification approach to improve the photocatalytic activity of TiO<sub>2</sub>. Se-Keun Park and Hyunho Shin<sup>14</sup> have explored the effect of HCl and H<sub>2</sub>SO<sub>4</sub> treatment on TiO<sub>2</sub> powder and its performance in the degradation of rhodamine B. They found

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that the acid treatment could enhance RhB adsorption on the  $TiO_2$  surface by protonation of the -COOH group via the Brønsted acid sites introduced by acid treatment. Sancun  $Hao^{15}$  and Kyung-Hee  $Park^{16}$  et al. reported that acid-treated  $TiO_2$  could improve the photovoltaic properties of dyesensitized solar cells by increasing the dye absorption and decreasing the charge-transfer resistance. Although acid-treated  $TiO_2$  has been applied in the domain of dye degradation and dye-sensitized solar cells, the application of acid-treated  $TiO_2$  in photocatalytic  $H_2$  production has never been investigated.

Herein, we applied an acid-treated P25 catalyst in  $\rm H_2$  photogeneration, expecting that the Brønsted acid site introduced by acid treatment could enhance the  $\rm H_2$  production by providing more protons. Moreover, plenty of oxygen vacancies and  $\rm Ti^{3+}$  sites can be introduced in P25 by acid treatment, which could capture electrons contributing to a higher carrier separation efficiency and hence a higher  $\rm H_2$  evolution rate during photoreaction. Such a simple strategy for enhanced photogeneration of  $\rm H_2$  has never been reported before.

# 2. Experimental

#### 2.1 Synthesis of HP25-60

HP25-*x* catalysts were prepared by acid treatment as follows: typically, 5 g P25 (≥99.8%, Sigma Aldrich) was immersed in 6 M hydrochloric acid (HCl, 36–38%, Sigma Aldrich) for 5

hours. Then the acid treated P25 was dried at 60 °C for 6 h and then heated at 60, 100, 200, and 300 °C for 3 h in an air atmosphere with a heating rate of 3 °C min<sup>-1</sup>. The samples obtained at different treatment temperatures are labeled as HP25-x, where x refers to the treatment temperature (x = 60, 100, 200 and 300 °C).

#### 2.2 Characterization

The crystal structure of the samples was observed using an X-ray diffractometer (XRD; Bruker D8 Advance X-ray diffractometer) at 40 kV and 40 mA equipped with a Cu anode X-ray tube (Cu K $\alpha$  X-rays,  $\lambda$  = 1.54056 Å). The morphology of the 3DOM Sr-TiO<sub>2</sub> sample was observed by scanning electron microscopy (SEM, Hitachi S-4800) and transmission electron microscopy (TEM, TALOS F200). The adsorption-desorption isotherms and the pore distributions of the catalysts were examined using volumetric adsorption equipment (NOVA Surface Area Analyzer Station A, USA). Ultraviolet visible (UV-vis) diffuse reflectance spectroscopy (DRS) was performed using a UV-vis spectrophotometer (TU-1901, China) in the range of 200-800 nm with BaSO<sub>4</sub> as the reference. The photoluminescence (PL) spectra were recorded using a Hitachi fluorescence spectrophotometer (F-7000) and the excitation wavelength was set at 264 nm.

#### 2.3 Photoelectrochemical measurements

Photoelectrochemical measurements were conducted with an Autolab PGSTAT302 N assembled with a 300 W Xe lamp in the frequency range of 10-100 kHz in a standard threeelectrode system using fabricated photocatalyst films as working electrodes, a Pt plate electrode as the counter electrode and an Hg/Hg2Cl2 (saturated in KCl) electrode as the reference electrode. The transient photocurrent response was measured in 0.5 mol L<sup>-1</sup> Na<sub>2</sub>SO<sub>4</sub> electrolyte with light on/ off cycles for different samples under visible light irradiation at a bias of 0 V vs. Hg/Hg<sub>2</sub>Cl<sub>2</sub>. To prepare the working electrode, the photocatalyst (4 mg) and 5 wt% Nafion solution (80 μL) were dispersed in isopropyl alcohol (920 μL) and sonicated for at least 1 h to form a homogeneous mixture. Then, the mixture (40 µL) was dropped cast on a 5 cm × 1 cm fluorine-doped tin oxide (FTO) glass electrode. Electrochemical impedance spectroscopy (EIS) tests were conducted in the same configuration at  $\eta = -0.6$  V (vs. RHE) from 10<sup>5</sup>-10<sup>-1</sup> Hz with an AC voltage of 20 mV. Mott-Schottky (M-S) plots were obtained at a frequency of 1000 Hz.

#### 2.4 Photocatalytic water splitting

The photocatalytic hydrogen evolution activities of different samples were tested by using a photocatalytic activity evaluation system (Labsolar-6A, Perfectlight, Beijing). Typically, 30 mg catalyst was added into an aqueous solution (90 mL) of triethanolamine (TEOA, 10 vol%, as a sacrificial reagent). 238 μL H<sub>2</sub>PtCl<sub>6</sub> solution (3 wt% Pt) was added into the above mixture, which was then exposed to light for 0.5 h to load Pt nanoparticles on the samples. Before irradiation, the whole system was sealed and vacuumed using a mechanical pump to eliminate any gas impurities. Then, the produced hydrogen was extracted every 1 hour with an online gas chromatograph (GC7900, Techcomp, Shanghai) equipped with a TCD and N2 carrier. A 300 W Xe lamp without any filters was used as the light source. The generated amount of hydrogen was evaluated according to the fitted standard curve. The apparent quantum yield (AQY) was calculated based on eqn (1). The light intensity was ca. 105.26 mW cm<sup>-2</sup> and the illuminated area for the reactor was about 28 cm<sup>2</sup>.

$$\begin{split} AQY(\%) &= \frac{number\ of\ reacted\ electrons}{number\ of\ incident\ photons} \times 100\% \\ &= \frac{number\ of\ evolved\ H_2\ molecules}{number\ of\ incident\ photons} \end{split} \tag{1}$$

# 3. Results and discussion

#### 3.1 Solid characterization

XRD was applied to detect the crystalline phases of P25 and HP25-x (Fig. 1). The sharp diffraction peaks at  $2\theta = 25.3^{\circ}$ , 36.9°, 37.8°, 38.6°, 48.0° and 53.9° are assigned to the (101), (103), (004), (112), (200) and (105) planes of anatase phase TiO<sub>2</sub> (JCPDS, No. 21-1272), respectively. And the sharp diffraction peaks at  $2\theta$  = 27.5°, 36.1°, 39.2°, 41.3°, 44.1° and 56.7° are ascribed to the (110), (101), (200), (111), (210) and (220) planes of rutile phase TiO<sub>2</sub> (JCPDS, No. 21-1276), respectively.18 Both the anatase and rutile phases are observed on untreated P25. In comparison to untreated P25, the diffraction of acid treated P25 shows no appreciable changes, indicating that the acid treatment does not destroy the crystal structure of P25. The XRD peaks decrease as the heating temperature increases. This suggests that the crystallinity of HP25-x decreases with the increase of treatment temperature.

To view the morphology of P25 and HP25-x (x = 60, 100), SEM images were obtained (Fig. 2). Before the acid treatment,

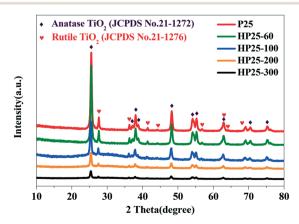


Fig. 1 XRD patterns of P25 and HP25-x (x = 60, 100, 200 and 300).

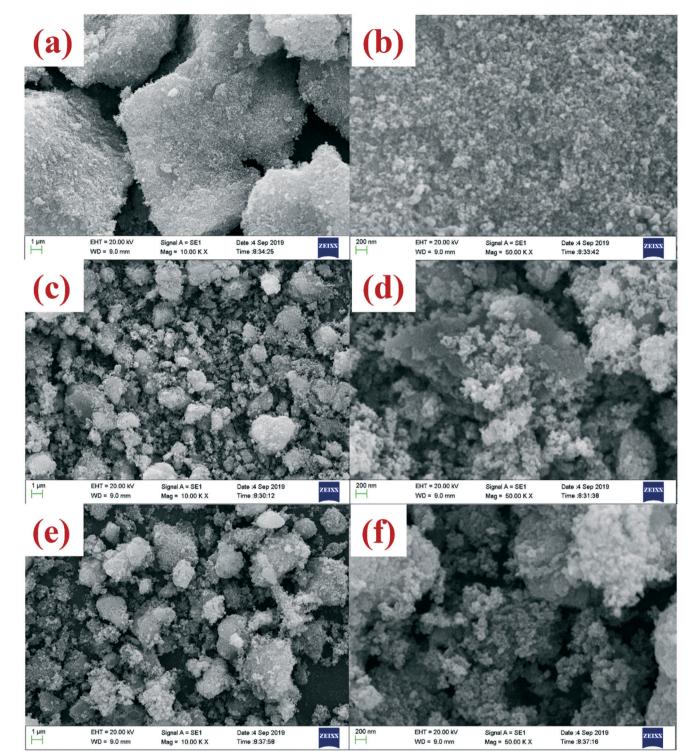


Fig. 2 SEM images of acid-treated P25 (a and b), HP25-60 (c and d) and HP25-100 (e and f).

many large particles of tens of microns were observed in the image of P25. After acid treatment, these large particles were etched into small irregular particles ranging from less than 1 μm to 3 μm. Because the morphologies of HP25-60 and HP25-100 are similar in SEM (Fig. 2), TEM was further carried out to obtain a more micro view of these samples (Fig. 3). The results show that HP25-60 and HP25-100 still

possess similar morphologies with small particles about 20-30 nm observed under high resolution TEM.

XPS measurements were also performed to elucidate the chemical valence state and the surface component of P25 and HP25-x (Fig. 4). The peaks located at 458.6 and 457.6 eV are assigned to  ${\rm Ti}^{4+}$   $2p_{3/2}$  and  ${\rm Ti}^{3+}$   $2p_{3/2}$ , respectively.<sup>19</sup> The Ti<sup>3+</sup> content in P25 is substantially increased after acid

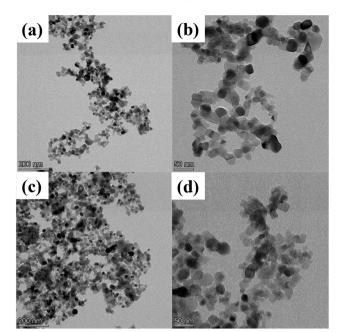


Fig. 3 TEM images of (a and b) P25-60 and (c and d) HP25-100.

treatment at 60 °C, which is consistent with previous reports<sup>16</sup> and is expected to broaden the light absorption scope by introducing an impurity level. 19 Once the treatment temperature increases to 100 °C, the Ti<sup>3+</sup> content decreases likely because the heating process in air favors the reoxidation of Ti3+ with O2.

The O peaks at 529.5 eV, 530.3 eV and 531.9 eV are ascribed to the O lattice (Ti<sup>4+</sup>-O), the O-atoms in the vicinity of oxygen vacancies (Ov, i.e. Ti3+O) and the OH bonds,19 respectively. The Ov content first increases after acid treatment at 60 °C, then decreases after treatment at 100 °C. The variation trend of O<sub>v</sub> content on the catalyst surface is consistent with that of Ti3+, which is quite reasonable since

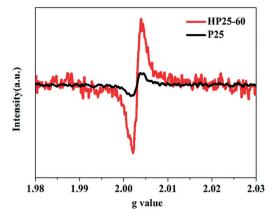


Fig. 5 ESR spectra of O<sub>v</sub> over P25 and HP25-60.

the O<sub>v</sub> formation is accompanied with the presence of Ti<sup>3+</sup>. Notably, the -OH groups are also substantially increased on the TiO<sub>2</sub> surface after acid treatment at 60 °C because TiO<sub>2</sub> tends to be protonated at acidic pH levels.20,21 Once the treatment temperature increases to 100 °C, the -OH groups significantly decrease due to the dehydroxylation effect at high temperature.

The existence of O<sub>v</sub> was further verified by ESR. In contrast to P25 that shows a small ESR peak, HP25-60 exhibits much steeper peaks at g = 2.003 (Fig. 5), which are assigned to electrons trapped by oxygen vacancies. 22-25 These results suggest that a large number of oxygen vacancies are generated on the surface of the acid-treated catalyst, consistent with the O<sub>v</sub> information obtained by XPS.

The FT-IR spectra of P25 and HP25-x (x = 60 and 100) samples are also obtained to characterize the surface -OH groups on P25 before and after acid treatment (Fig. 6). The bands at 3420 cm<sup>-1</sup> and 1632 cm<sup>-1</sup> are assigned to the stretching vibrations and the bending vibrations of adsorbed water, respectively. 14,26 The bands at 3640 cm and 1260-

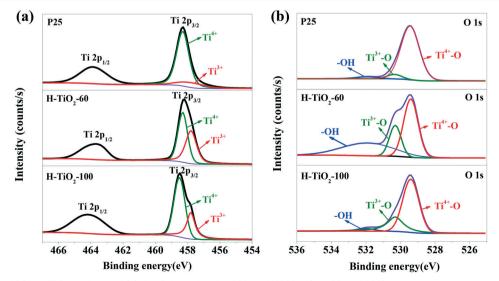


Fig. 4 XPS spectra of (a) the Ti 2p region and (b) the O 1s region for P25 and HP25-x (x = 60 and 100).

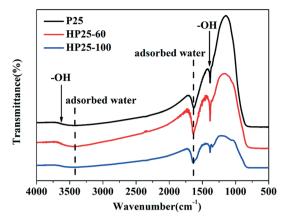


Fig. 6 FT-IR spectra of P25 and HP25-x (x = 60 and 100).

1350 cm<sup>-1</sup> are assigned to the stretching vibrations and the bending vibrations of -OH groups, respectively.<sup>27-29</sup> Notably, the amount of adsorbed water and -OH groups both increased substantially on the P25 surface after acid treatment at 60 °C. Once the treatment temperature further increased to 100 °C, the amount of adsorbed water and -OH groups both decreased significantly. These results are consistent with the information obtained from the XPS spectra of the O 1s region.

#### 3.2 Specific surface area analysis

The N<sub>2</sub> adsorption-desorption isotherms of these samples are shown in Fig. 7 and Table 1. All the samples present a type IV isotherm with a typical H<sub>3</sub> hysteresis loop, indicating the existence of a pore structure.<sup>17</sup> After acid treatment, the surface area and pore volume were substantially increased due to the etching effect of acid on TiO2. As the treatment temperature increases, small changes are observed on the surface area and pore volume, which indicates that the treatment temperature exerts a small influence on the textural properties of the catalysts.

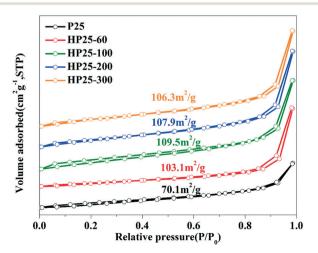


Fig. 7 Nitrogen adsorption-desorption isotherm plots of P25 and HP25-x (x = 60, 100, 200and 300).

Table 1 The summary of nitrogen adsorption-desorption isotherm relevant parameters of P25 and HP25-x (x = 60, 100, 200 and 300)

Catalysis	Surface area $(m^2 g^{-1})$	Pore volume	Pore diameter
P25	70.1	0.236	3.418
HP25-60	103.1	0.423	3.449
HP25-100	109.5	0.462	3.428
HP25-200	107.9	0.499	3.452
HP25-300	106.3	0.518	3.443

#### 3.3 Optical properties

UV-vis DRS provides information about the optical properties of the prepared samples (Fig. 8a). HP25-x and P25 show similar light absorbance in the range of 200-400 nm. Specifically, in comparison to untreated P25, acid treated HP25-x samples show a slight red shift in the light absorption range, likely due to the formation of an impurity level between the CB and VB caused by the presence of Ti<sup>3+</sup> (Fig. 4).30

The optical band gap of HP25-x and P25 was calculated by using  $(\alpha h v)^n = A(h v - Eg)$  (Fig. 8b), where  $\alpha$  is the absorption coefficient, A is the parameter that is related to the effective masses associated with the valence and conduction bands, nis 1/2 for a direct transition, hv is the absorption energy, and Eg is the band gap energy, respectively.31 The band gaps for P25 and HP25-x (x = 60, 100, 200 and 300) were determined to be 3.15, 3.10, 3.07, 3.11 and 3.11 eV, respectively. This indicates that the band gap changes caused by acid treatment are kind of small.

#### 3.4 Photoelectrochemical analysis

Photocurrent measurement is a technique used to examine the separation rate and transfer efficiency of photointroduced carriers in a catalytic process<sup>32</sup> (Fig. 9a). In general, the higher photocurrent intensity suggests a higher carrier separation efficiency. The photocurrent sharply increased under light illumination, and simultaneously decayed as the light turned off. This suggests that electrons are excited on the sample surface and transferred to the electrode to generate photo-induced current under light irradiation. HP25-60 shows a much higher photocurrent intensity than the other samples, indicating a much higher carrier separation and transfer efficiency introduced by acid treatment. This is likely because more electrons can be captured by the oxygen vacancies in HP25-60, and more holes can be captured by the Ti<sup>3+</sup> sites.<sup>19,33</sup> As the treatment temperature increases from 100 to 300 °C, the photocurrent density response decreases substantially likely because the Ti<sup>3+</sup> sites were re-oxidized into Ti<sup>4+</sup> sites at higher temperature (Fig. 4).

EIS is also an effective method for probing the charge transfer efficiency at the interface of modified electrodes.3 EIS plots of P25 and HP25-x (x = 60, 100, 200 and 300) are displayed in Fig. 9b. The arc radius in the Nyquist plot of HP25-60 is the smallest, indicating that HP25-60 has the

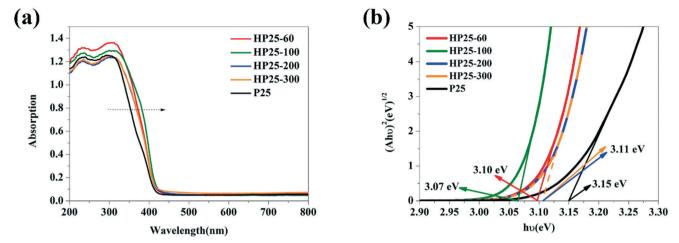


Fig. 8 (a) UV-vis spectra and (b) band gap energy of P25 and HP25-x (x = 60, 100, 200 and 300).

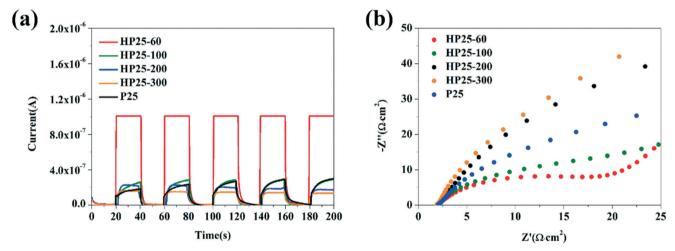


Fig. 9 (a) J-V curves plotting the photocurrent density as a function of the applied potential. (b) EIS spectra of P25 and HP25-x (x = 60, 100, 200 and 300) in 0.5 M NaOH ethanol solution.

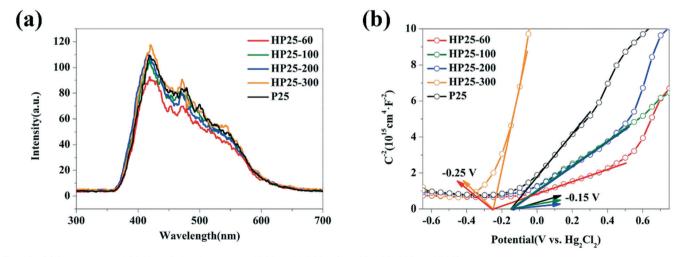


Fig. 10 (a) PL spectra and (b) Mott–Schottky curves of P25 and HP25-x (x = 60, 100, 200 and 300).

**Table 2** The donor density of P25 and HP25-x (x = 60, 100, 200 and

Catalysis	The donor density	
P25	1.22 × 10 <sup>16</sup>	
HP25-60	$2.35 \times 10^{17}$	
HP25-100	$1.19 \times 10^{17}$	
HP25-200	$1.09 \times 10^{17}$	
HP25-300	$6.07 \times 10^{15}$	

slowest charge recombination and fastest charge separation, which is consistent with the conclusions obtained from the photocurrent (Fig. 9a).

Photoluminescence (PL) spectroscopy is also an effective technique to study the electronic structure, optical and photochemical properties of semiconductor materials, by which the efficiency of carrier separation can be obtained.<sup>34</sup> The higher the emission peak intensity in the PL spectra, the higher the carrier recombination efficiency. The main emission peaks of HP25-x and P25 appear in the range of 400-600 nm (Fig. 10a). The lowest peak intensity of HP25-60 (Fig. 10a) suggests that HP25-60 possesses the lowest carrier recombination, i.e. the highest carrier separation efficiently among all the samples, which is consistent with the conclusions obtained from the photocurrent and impedance characterization.

To better understand the band structure and charge separation of HP25-x and P25, Mott-Schottky analysis was applied (Fig. 10b) and the donor density was calculated using the Mott-Schottky relation [eqn (2)]:

$$\frac{1}{C_{sc}^{2}} = \frac{2}{e\varepsilon\varepsilon_{0}N} \left( E_{appl} - E_{fb} \right) - \frac{KT}{e} \tag{2}$$

where  $C_{\rm sc}$  is the capacitance of the space-charge layer, e is the charge of the electron,  $\varepsilon$  is the dielectric constant of the semiconductor,  $\varepsilon_0$  is the vacuum permittivity of free space, N is the donor density (electron donor concentration for an n-type semiconductor or hole acceptor concentration for a p-type semiconductor),  $E_{appl}$  is the applied potential,  $E_{fb}$  is the flat-band potential, K is the Boltzmann constant, and T is the absolute temperature.<sup>35</sup> It is generally believed that  $E_{\rm fb}$  is 0.1 V below the conduction band position  $(E_{CB})$  for n-type semiconductors.<sup>36</sup> The flat-band potentials  $(E_{fb})$  were calculated and are summarized in Table S1.† The donor density of HP25-60 is 6.8 times higher than that of P25 (Table 2) and is the highest among all the samples which is in good agreement with the photocurrent and EIS results.

#### 3.5 Photocatalytic performance

The H<sub>2</sub> evolution capability over P25 and HP25-x was tested with (Fig. 11) or without (Fig. S1†) Pt as a co-catalyst. After 5 h of reaction, the H<sub>2</sub> production over HP25-60 reached 4852.9 or 1229.2 μmol h<sup>-1</sup> with or without a Pt co-catalyst, which is 7.1 or 5.3 times higher than 232.3  $\mu$ mol h<sup>-1</sup> of P25. The AOY values over HP25-60 with or without a Pt co-catalyst are 21.3% or 5.4%, respectively. However, as the treatment temperature increases from 60 °C to 100, 200 and 300 °C, the hydrogen production decreases significantly, with HP25-200 even producing a lower amount of H2 than untreated P25. The decrease of hydrogen production at higher acidtreatment temperature can be attributed to the simultaneous decrease of -OH, Ti3+ and oxygen vacancies at higher treatment temperature (Fig. 4). On the one hand, the decrease of -OH groups results in a lower amount of Brønsted acid sites, and hence fewer protons for H2 evolution. On the other hand, the decrease of Ti<sup>3+</sup> and oxygen vacancies could result in a lower efficiency of carrier separation and transfer, leading to lower hydrogen production.

#### 3.6 Possible photocatalytic mechanism

Based on the above results, a possible mechanism for the enhanced photocatalytic H2 production efficiency of HP25-x is proposed (Fig. 12). Among all investigated catalysts, HP25-60 has the optimal activity in H2 photogeneration. After acid treatment,

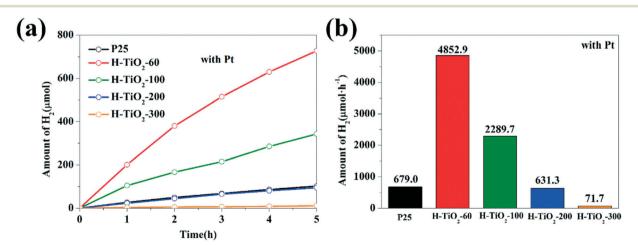


Fig. 11 (a and b) Photocatalytic capacity for H<sub>2</sub> production from water splitting over P25 and HP25-x (x = 60, 100, 200 and 300) under the conditions with Pt as a co-catalyst.

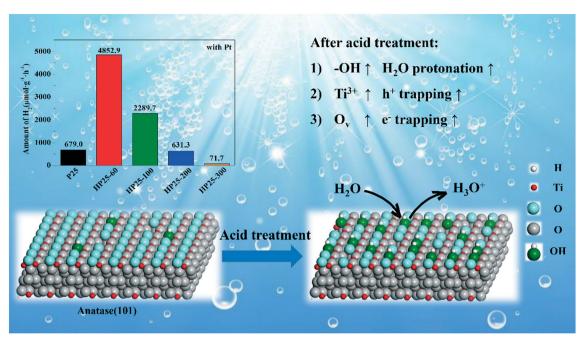
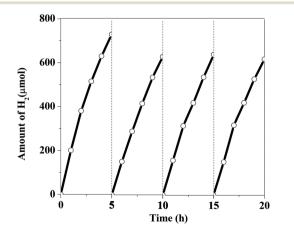


Fig. 12 Possible photocatalytic mechanism of photocatalytic H<sub>2</sub> production over HP25-60 under simulated light.

the contents of -OH, Ti3+ and oxygen vacancies increase significantly (Fig. 4). On the one hand, the substantial amount of -OH groups introduced by acid treatment could act as Brønsted acid sites enhancing the water adsorption and providing protons for H<sub>2</sub> evolution. 14 On the other hand, after acid treatment, more electrons can be captured by more oxygen vacancies in HP25-60, and more holes can be captured by more Ti3+ sites, resulting in an enhanced carrier separation and transfer efficiency. Although the red shift of the light absorption scope can be observed, such a slight shift introduced by acid treatment exerts a faint influence on the band gap and the light absorption capability of P25, which suggests that this effect can be removed from the reasons for enhanced H<sub>2</sub> production. To sum up, the enhanced photocatalytic H<sub>2</sub> production activity of HP25-60 is mainly attributed to the co-effect of Brønsted acid sites, Ti3+ and oxygen vacancies. Such a modification strategy can be used as a



**Fig. 13** Repeated photocatalytic hydrogen production experiments over HP25-60.

universally applicable method for other valence-alternative metal oxide semiconductors for H<sub>2</sub> photogeneration.

#### 3.7 Stability evaluation

Photocatalyst recycle experiments of HP25-60 were carried out to examine the catalyst stability (Fig. 13). The recycled HP25-60 still shows a high photocatalytic  $H_2$  production rate after 4 cycles, indicating that the acid treated P25 possesses excellent stability during  $H_2$  evolution experiments.

## 4. Conclusions

In this paper, highly efficient P25 catalysts for photocatalytic H<sub>2</sub> evolution were successfully fabricated via a simple acid treatment (HP25-x, x = 60, 100, 200 and 300). After 5 h of reaction, the H<sub>2</sub> production over HP25-60 reached 4852.9 or 1229.2 μmol h<sup>-1</sup> with or without a Pt co-catalyst, respectively, which is 7.1 or 5.3 times higher than 232.3  $\mu$ mol h<sup>-1</sup> of P25. The enhanced photocatalytic H2 production activity of HP25-60 is due to the presence of substantial Brønsted acid sites, as well as Ti3+ sites and oxygen vacancies. The Brønsted acid sites can protonate H2O, hence enhancing H2O adsorption and providing protons for H<sub>2</sub> production. The Ti<sup>3+</sup> sites and oxygen vacancies can suppress carrier recombination and promote carrier transfer efficiency by trapping electrons and holes, respectively. Such a simple acid treatment strategy can be used as a universally applicable method for modification of other photocatalysts (e.g. valence-alternative metal oxide semiconductors) for enhanced H2 photogeneration.

### Conflicts of interest

There are no conflicts to declare.

# Acknowledgements

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