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# Boosting the circularly polarized luminescence of small organic molecules *via* multi-dimensional morphology control†

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Achieving a higher dissymmetry factor is a crucial issue in developing circularly polarized luminescence (CPL) materials. Here, by tailoring the solvent composition and the morphology of the same chiral emissive small molecules (*R*- or *S*-SPAN), circularly polarized emission with a boosted dissymmetry factor (two orders) was realized. It was found that by regulating the water fraction in the mixed THF/H<sub>2</sub>O, we were able to achieve kinetic control over association of chiral emissive *R*- or *S*-SPAN into various nanostructures with 0D nanospheres, 2D nanoflakes and 3D stacked nanoflakes. These nanostructures are all CPL active. Remarkably, the dissymmetry factors of the nanostructures were significantly enhanced compared to those of the molecules and further boosted in different morphologies, from  $\sim 10^{-4}$  (0D nanospheres) to  $10^{-3}$  (2D flake) to  $\sim 10^{-2}$  (3D nanoflakes). The enlarged  $g_{lum}$  value could be assigned to a good packing induced strong luminescence of an excimer. This strategy provides an efficient way to fabricate higher dissymmetry factor CPL organic nanomaterials by only changing the supramolecular architectures while using the same chiral small molecules.

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## Introduction

Chirality is a basic characteristic of nature which can be incisively and vividly exemplified by molecular, macromolecular or supramolecular levels of bioorganic systems, such as amino acids, enzymes, proteins, sugars, RNA and DNA.<sup>1–8</sup> In recent years, excited state chirality, namely circularly polarized luminescence (CPL), has aroused much attention due to its potential for application in many fields.<sup>9–20</sup> CPL originates from giving priority to one handed circularly polarized emission and the level of polarization is usually characterized by the luminescence dissymmetry factor ( $g_{lum}$ ). The value of  $g_{lum}$  ranges from  $-2$  to  $+2$  and the maximum value of  $|g_{lum}|$  represents completely left or right-handed circularly polarized light.<sup>21–23</sup> To broaden the application of CPL-active materials, it is necessary to pursue large  $g_{lum}$  of chiral luminescent systems. To date, the

largest  $g_{lum}$  value with about 1.3 was obtained in lanthanide complexes, that is cesium tetrakis(3-heptafluoro-butylryl-(+)-camphorato) Eu(III) complexes.<sup>24</sup> On the other hand, in contrast to lanthanide complexes, research on simple organic molecules with CPL activity has become popular due to the inherent advantages of organic luminescent molecules.<sup>25,26</sup> Compared with lanthanide complexes, the emission of pure organic CPL-active materials could be tuned more flexibly by regulating the electronic levels of the excited state. For instance, many strategies have been realized to modulate the electronic state of organic molecules, such as by chemical transformations of substituents, self-assembly and by changing the environment through external stimuli, *e.g.* temperature, pH, concentration, light, magnetism and solvents.<sup>27–38</sup> Unfortunately, the  $g_{lum}$  value of organic systems is relatively low and generally falls in the range of  $10^{-5}$  to  $10^{-3}$ , which precludes them from being the best candidates for CPL research. To amplify the luminescence dissymmetry factors, different approaches have been adopted, which include the formation of receptor-ion complexes, configurational changes upon binding with guest ions or molecules and self-assembly of chiral molecules.<sup>39–46</sup> For example, Tang and co-workers, by introducing the concept of aggregation-induced emission, realized the amplification of  $g_{lum}$ , which could be regarded as an excellent approach for fabricating efficient CPL-active organic solid materials.<sup>25,47,48</sup> Very recently, we have demonstrated that energy transfer, including Förster resonance energy transfer and the photon upconversion process, could remarkably amplify the  $g_{lum}$

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**Fig. 1** (a) Fluorescence spectra of *R*-SPAN aggregates in various  $f_w$  ( $\lambda_{ex} = 320$  nm; intensity normalized at a maximum value). (b) CD spectra of *R*- and *S*-SPAN in pure THF. (c) CPL spectra of *R*-/*S*-SPAN in pure THF. (d) Dissymmetry factor  $g_{lum}$  of *R*-SPAN aggregates in various  $f_w$ . Upon increasing the volume fraction of water, the maximum  $g_{lum}$  for  $f_w$  90% was amplified two orders of magnitude in comparison with the  $g_{lum}$  for  $f_w$  0% and 50%. ( $[R\text{-SPAN}] = 1.5$  mM,  $\lambda_{ex} = 320$  nm).

could significantly amplify the  $g_{CD}$  by controlling the solvents. Thus, by controlling the morphology of nanostructures aggregated at various water fractions, amplified supramolecular chirality was obtained. The amplified  $g_{CD}$  laid the foundation for enhanced CPL as discussed below. The CPL spectra could provide more direct evidence for chiral dissymmetry in the fluorescence of the monomer and aggregates (Fig. 1c and S5†). Upon increasing the volume fraction of water, the intensity of the CPL was obviously enhanced. The extent of chiral

dissymmetry in fluorescence is quantified using the anisotropy factor,  $g_{lum}$ , of CPL, which is given by the equation  $g_{lum} = 2(I_L - I_R)/(I_L + I_R)$ , where  $I_L$  and  $I_R$  are the intensities of the left- and right-handed circularly polarized emissions, respectively.<sup>60,61</sup> As shown in Fig. 1d, in pure THF solution and  $f_w$  50%, the  $g_{lum}$  values were about  $2.1 \times 10^{-4}$  and  $2.4 \times 10^{-4}$ , respectively. Upon increasing the volume fraction of water, the  $g_{lum}$  value increased by an order of magnitude. The maximum  $g_{lum}$  values for  $f_w$  85% and 90% were  $7.2 \times 10^{-3}$  and  $2.9 \times 10^{-2}$ , respectively. The



**Fig. 2** SEM images of nanostructures of *R*-SPAN in different  $f_w$ : (a and b) 50%, (c and d) 85% and (e and f) 90%, respectively. ( $[R\text{-SPAN}] = 1.5$  mM, scale bar 1  $\mu$ m). Upon increasing the fraction of water, the morphologies transformed from 0D nanospheres with a rough surface or with a hole to 2D and finally to 3D layered nanoflakes.



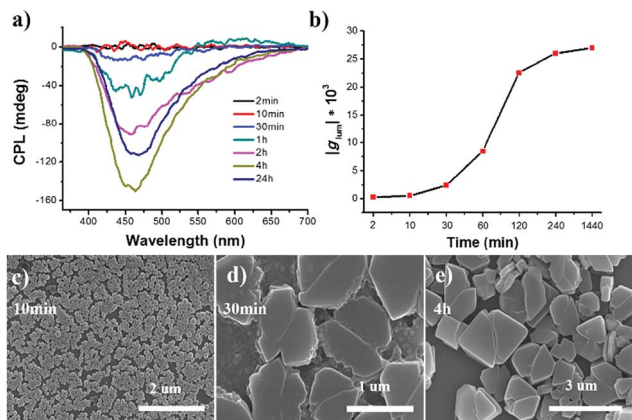


Fig. 3 (a) Aging time-dependent CPL spectra of *R*-SPAN nanostructures in the water fraction of 90%. (b) The plot of  $g_{lum}$  value of nanostructures recorded at different times. Time-dependent SEM images of *R*-SPAN nanostructures in the water fraction of 90% at (c) 10 min, (d) 30 min and (e) 4 h, respectively. ( $[R\text{-SPAN}] = 1.5 \text{ mM}$ ,  $\lambda_{ex} = 320 \text{ nm}$ ).

obtained  $g_{lum}$  value for  $f_w$  90% in the pure organic systems was a relatively high value. The  $g_{lum}$  values of *S*-SPAN aggregated at various water fractions are shown in Fig. S6.† The order of magnitude of  $g_{lum}$  values of *S*-SPAN nanostructures at different water fractions was the same as that of *R*-SPAN. The obvious amplified  $g_{lum}$  values may have resulted from the strong intermolecular coupling in the nanostructures.

Scanning electron microscopy (SEM) measurement was employed to investigate the morphological transformation of aggregates under different conditions. As shown in Fig. S7a,† the *R*-SPAN molecule formed amorphous structures upon evaporation of THF. When the water fraction reached 50%, two kinds of nanoparticles with a rough surface and with a hole could be observed (Fig. 2a and b). Upon increasing the fraction of water to 85%, two-dimensional nanoflakes were observed (Fig. 2c and d). Finally, when the water fraction reached 90%, three-dimensionally stacked flakes, which exhibit the largest  $g_{lum}$  value (Fig. 2e and f), could be obtained. The morphological transformation of *S*-SPAN nanostructures at various water fractions is shown in Fig. S6.† Thus, by varying the composition of the solvent, a dramatically changed CPL dependent on the morphological transformation could be obtained. We also tried higher volume fractions of water, such as 96% and 98%. Unlike the sample with a water fraction of 90%, the one with  $f_w$  96% exhibited the structure of nanobelts, while hollow microspheres were observed in  $f_w$  98% (Fig. S7b and c†). However, it is hard to get a plausible CPL signal under these conditions. This was due to the weak luminescence at a higher volume fraction of water.<sup>62</sup>

The ripening process of *R*-SPAN in the water fraction of 90% was monitored by CPL measurement, as shown in Fig. 3a. The time-dependent CPL investigations in the water fraction of 90% showed that the CPL intensity dramatically increased. The time-dependent emission was also investigated, and it showed a similar tendency to CPL, as shown in Fig. S8.†

The  $g_{lum}$  value at different ripening times is shown in Fig. 3b. In the first two minutes, the  $g_{lum}$  value of the obtained sample



Fig. 4 (a) XRD patterns and (b) FT-IR spectra of self-assembled *R*-SPAN nanostructures formed in various mixed solvents. The drop-cast film of *R*-SPAN solution (THF) was used as 0% for these tests.  $[R\text{-SPAN}] = 1.5 \text{ mM}$ . The diffraction peaks of the water fractions of 85% and 90% showed a body-centred cubic packing of the *R*-SPAN molecules. Upon increasing the water fraction from 0% to 90%, the FT-IR peaks ascribed to the stretching vibration of hydroxyl obviously shifted toward lower wavenumbers.

was  $2.2 \times 10^{-4}$ . After 4 hour ripening, the  $g_{lum}$  value reached  $2.6 \times 10^{-2}$ , and remained at an almost constant value, confirmed by testing the sample after 24 hours ( $2.7 \times 10^{-2}$ ). SEM was carried out to carefully investigate the morphological transformation at different ripening times. In the first 10 minutes, mono-disperse nanoparticles with a size distribution of about 140 nm were obtained, as shown in Fig. 3c. When the ripening time reached 30 minutes, the nanoparticles stacked together and fused to form a laminated flake structure with rough edges (Fig. 3d). After 4 hours, stacked 3D nanoflakes with a smooth surface were observed (Fig. 3e). Accompanying the morphology transformation from 0D nanospheres to 3D nanoflakes, the CPL activity also showed amplification. The  $g_{lum}$  value was amplified by two orders of magnitude to 0.027. The enhanced CPL activity of time-dependent morphology transformation in  $f_w$  90% suggested that the CPL activity exhibited morphological dependence.

To further clarify the formation of nanostructures with gradually enhanced CPL, X-ray diffraction (XRD) was carried out, as





Fig. 5 Molecular packing analysis results and speculated mechanism for the amplification of  $g_{lum}$ . (a) Theoretically predicted growth morphology of an *S*-SPAN crystal based on the attachment energies calculated with Material Studio package. (b) Crystal packaging along the vertical growth direction. (c) Speculated mechanism for the amplification of  $g_{lum}$ .

shown in Fig. 4a. The diffraction pattern of the *R*-SPAN cast film exhibited only one broad peak which indicated the formation of amorphous structures. For the  $f_w$  50% sample, three diffraction peaks were observed at  $2\theta$  values of  $7.01^\circ$ ,  $7.37^\circ$  and  $21.94^\circ$  with  $d$  spacings of 1.26 nm, 1.19 nm, and 0.41 nm, respectively. Clearly, upon increasing the volume fraction of water to 85% and 90%, three new diffraction peaks were observed at  $2\theta$  values of  $10.46^\circ$ ,  $11.71^\circ$  and  $16.55^\circ$ . The whole diffraction peaks of  $f_w$  90% gave  $d$  spacings of 1.20, 0.85, 0.75 and 0.54 nm with a  $d$  spacing ratio of about  $1 : \sqrt{2} : \sqrt{3} : \sqrt{5}$ , indicating a body-centred cubic packing of the molecules.<sup>63</sup> The first order diffraction peak of 3D nanoflakes was located at a  $2\theta$  value of  $7.34^\circ$  while the peak of 2D nanoflakes was located at a  $2\theta$  value of  $7.24^\circ$ , which indicated a closer molecular packing of the 3D nanoflakes than the 2D ones. In addition, the selected area electron diffraction (SAED) of the nanostructures obtained at  $f_w$  85% and 90% showed ordered diffraction patterns (Fig. S9 and Table S2†). The obtained inter-lattice spacing was about 0.53–0.54 nm, which could be estimated from the result of XRD. These results suggested that the obtained 2D and 3D nanoflakes had ordered molecular packing and a crystalline nature to some extent.

To gain further insight into the non-covalent interactions, Fourier transform infrared (FT-IR) spectra of the nanostructures were obtained and are shown in Fig. 4b. The peaks located at 885 and  $846\text{ cm}^{-1}$  could be attributed to the P–O stretching vibration

(aromatic ring). The spectra with sharp peaks at around  $1214\text{ cm}^{-1}$  could be ascribed to the P=O stretching vibration. In addition, the peak appearing at  $1442\text{ cm}^{-1}$  could be ascribed to the stretching vibration of  $-\text{CH}_2$ . The band appearing at  $3052\text{ cm}^{-1}$  could be ascribed to the stretching vibration of unsaturated C–H single bonds.<sup>64</sup> Upon increasing the fraction of water, the stretching vibration of the hydroxyl group of the phosphate moiety shifted toward lower wavenumbers, from  $3427\text{ cm}^{-1}$  ( $f_w$  0%) to  $3382\text{ cm}^{-1}$  ( $f_w$  90%). This was indicative of the existence of hydrogen bonding between intermolecular hydroxyl groups. Simultaneously, the peak became broader, which suggested the existence of an enhanced hydrogen bonding. These results indicated that the enhanced intermolecular hydrogen bond might be the main reason for enhancing the molecular stacking of 2D and 3D nanoflakes with relatively higher  $g_{lum}$  values.<sup>65</sup>

To gain a deep insight into the CPL amplification upon increasing the fraction of water, molecular packing analyses were carried out. We applied Materials Studio as a simulation tool for theoretically predicting the growth morphology of the *S*-SPAN single crystal. The simulation was performed using MS's morphology component based on the attachment energies. Firstly, the single-crystal of *S*-SPAN was incubated with THF/water mixed solvent. As shown in Fig. 5a, the crystal morphology of SPAN was plate-like, which was in agreement with the SEM results. The detailed crystallographic data are



shown in Table S3.† Based on the XRD patterns, the growth in the thickness of the microplates was along the [001] direction, which could be attributed to the molecular packing based on  $\pi$ - $\pi$  interaction of anthracene groups and hydrogen bonding between intermolecular hydroxyl groups as shown in Fig. 5b. With all of this in mind, we could speculate on a plausible mechanism for the amplification of CPL dissymmetry factor ( $g_{lum}$ ). When the water fraction was relatively low, the SPAN molecules aggregated in the form of microspheres, which was an amorphous packing mode and showed a relatively low  $g_{lum}$  ( $\sim 10^{-4}$ ). Upon increasing the water content, SPAN molecules could aggregate in the form of microplates in crystalline packing mode, and the organized packing could amplify the CPL dissymmetry factor ( $g_{lum} \sim 10^{-3}$ ). Furthermore, when the water fraction was elevated to a higher level, more crystal units packed in the thickness direction, *i.e.*, more ordered structures, were involved in the CPL generation process. Thus,  $g_{lum}$  could be further amplified ( $g_{lum} \sim 10^{-2}$ ).

## Conclusions

In summary, by tailoring the composition of the solvent (THF/H<sub>2</sub>O), the same chiral emitter *R*- or *S*-SPAN could be constructed into various nanostructures, including 0D nanospheres, 2D and 3D nanoflakes. Accompanying the morphological transformation, an amplified circularly polarized emission is observed. The maximum  $g_{lum}$  value could reach 0.029, which is a relatively large value in pure organic systems. The strong intermolecular interaction resulted in the formation of a more orderly and compact arrangement of nanostructures. The good packing of nanostructures induced a strong excimer emission, which could contribute a large  $g_{lum}$  value. The morphological dependence of emissive nanostructures on CPL activity can lead to controlled modulation of chiroptical properties, offering great potential for fabricating chiroptical organic nanomaterials.

## Experimental

### Materials

(11aS) and (11ab)-3,7-Di-9-anthracenyl-10,11,12,13-tetrahydro-5-hydroxy-5-oxide diindeno[7,1*de*:1',7'-*fg*][1,3,2] dioxaphosphocin (*S*-SPAN and *R*-SPAN) were purchased from Daicel Chiral Compounds (Shang Hai) without further purification. Tetrahydrofuran was purchased from TCI. Milli-Q water (18.2 MO cm) was used in all cases.

### Preparation

*S*-SPAN and *R*-SPAN were dissolved in tetrahydrofuran (THF) with a volume of 1 mL, 500  $\mu$ L, 150  $\mu$ L and 100  $\mu$ L, respectively. And then 500  $\mu$ L, 850  $\mu$ L and 900  $\mu$ L of water were added to the latter three. Thus, the nanostructures with various morphologies could be obtained. The final concentration of *R*- or *S*-SPAN of all samples was 1.5 mM.

## Characterization

UV-vis spectra, fluorescence spectra and CD spectra were obtained using a Hitachi UV-3900, Zolix Omin- $\lambda$ 500i monochromator with a photomultiplier tube PMTH-R 928 and JASCO J-810 spectrometers, respectively. CPL measurements were performed with a JASCO CPL-200 spectrometer. XRD analysis was performed on a Rigaku D/Max-2500X-ray diffractometer (Japan) with CuK $\alpha$  radiation ( $\lambda = 1.5406$  Å), operated at a voltage of 40 kV and a current of 200 mA. FTIR studies were performed with a JASCO FTIR-660 spectrometer. SEM was performed on a Hitachi S-4800FE-SEM with an accelerating voltage of 10 kV. TEM and selected area electron diffraction were performed on a transmission electron microscope, JEM1011. The fluorescence lifetime measurements were recorded on an Edinburg FLS-980 fluorescence spectrometer using time-correlated single photon counting.

## Conflicts of interest

There are no conflicts to declare.

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