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Synergistic effects of metallic and non-metallic element-doped electrocatalysts for the hydrogen evolution reaction: a review

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The hydrogen evolution reaction (HER) is a key electrochemical process for sustainable hydrogen production via water splitting. However, its practical implementation is hindered by sluggish reaction kinetics and reliance on noble metal catalysts like platinum, which are costly and scarce. To overcome these limitations, synergistic doping of metallic (e.g., Fe, Co, Ni, Mo, Mn, Cu, Pt) and non-metallic (e.g., P, N, B, O, S) elements has emerged as an effective strategy to enhance catalytic activity. This dual-doping approach enables fine-tuning of the catalyst's electronic environment, increases active site density, and improves hydrogen adsorption/desorption behaviour. Metallic dopants modulate conductivity and the electronic structure of active sites, while non-metallic heteroatoms introduce charge redistribution, surface defects, and chemical polarity – collectively accelerating HER kinetics. This review critically examines recent advances in the synthesis and performance of heteroatom-doped HER electrocatalysts under acidic and alkaline conditions. Emphasis is placed on how compositional tuning, structural design, and interface engineering contribute to improved catalytic performance, including low overpotentials, favourable Tafel slopes, and long-term stability. These developments underscore the potential of heteroatom doping as a versatile platform for designing next-generation HER catalysts for scalable and economically viable hydrogen energy systems.

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1. Introduction

The global transition toward sustainable energy systems has intensified the search for alternative energy carriers capable of replacing fossil fuels, which are increasingly associated with



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environmental degradation, climate change, and geopolitical instability.¹ Among the various candidates, hydrogen (H₂) has emerged as a highly promising energy vector due to its high gravimetric energy density, carbon-free combustion, and broad applicability in transportation, chemical synthesis, fuel cells, and energy storage systems.^{2,3} Electrocatalytic hydrogen production *via* water splitting offers a clean and scalable route to H₂ generation, especially when driven by renewable electricity sources such as solar, wind, or tidal energy.^{4,5} Within this process, the HER is a critical half-reaction that governs the overall efficiency and viability of water electrolysis systems. Despite its simplicity in stoichiometry, the HER involves complex electron–proton transfer steps that are highly dependent on the energetics of the catalytic surface.^{6,7} Platinum-group metals (PGMs) are widely regarded as the benchmark HER catalysts due to their near-zero hydrogen adsorption free energy (ΔG_{H^*}) and exceptional exchange current densities.⁸ However, the scarcity, high cost, and limited durability of these noble metals have posed significant barriers to their large-scale deployment in practical systems.⁹ As a result, extensive research efforts have been directed toward developing Earth-abundant, cost-effective, and efficient non-precious metal-based HER catalysts.^{10,11} One of the most promising approaches in this domain is heteroatom doping, where foreign atoms, either metallic or non-metallic, are intentionally introduced into the host lattice of a catalytic material to modulate its physico-chemical properties (Fig. 1). Doping enables the tuning of electronic structures, creation of defect states, optimization of hydrogen adsorption sites, and enhancement of charge transfer kinetics.¹² Specifically, metallic dopants, such as 3d transition metals (*e.g.*, Fe, Co, Ni, Mn, Cu), can significantly alter the d-band center of the host material, thereby adjusting the hydrogen binding energy and improving the intrinsic catalytic activity.¹³ On the other hand, non-metallic dopants such as nitrogen (N), phosphorus (P), sulfur (S), boron (B), and oxygen (O), play an essential role in inducing charge redistribution, defect engineering, and improving hydrophilicity and conductivity.¹⁴ Recent studies have demonstrated that synergistic co-doping of both metallic and non-metallic elements can yield

even greater improvements in HER activity than single-element doping strategies.¹⁵ This synergistic enhancement arises from the complementary effects of the two dopants: while metallic dopants optimize the electronic configuration of active sites, non-metallic elements further modulate the local charge environment, induce lattice strain, and stabilize catalytic intermediates.^{14,16} For example, in carbon-based materials,¹⁷ the dual incorporation of transition metals with N or P not only creates M–N_x or M–P_x moieties with high activity but also enhances the electron delocalization throughout the matrix, leading to improved conductivity and reaction kinetics.¹⁸ Moreover, synergistic doping can influence the thermodynamics of hydrogen adsorption and desorption, tuning the ΔG_{H^*} toward thermoneutrality (~ 0 eV), which is a key criterion based on the volcano plot relationship for optimal HER catalysts. In transition metal chalcogen transport, and phosphides, co-doping can promote phase transformation, enhance interfacial charge transport, and increase the electrochemically active surface area.¹⁹ Additionally, co-doped systems often exhibit superior durability under harsh electrochemical conditions, attributable to stabilized bonding configurations and suppressed leaching of active species. Despite significant progress, the rational design of synergistically doped HER catalysts remains a complex challenge. Critical aspects such as dopant selection, atomic configuration, dopant–host interaction, and synthetic controllability need to be systematically investigated. Advanced characterization techniques such as X-ray absorption spectroscopy (XAS), transmission electron microscopy (TEM), and density functional theory (DFT) simulations are increasingly being employed to gain atomistic-level insights into the role played by dopants and their interactions with the host material.

This review aims to provide a comprehensive overview of recent advancements in the development of synergistically doped metallic and non-metallic HER electrocatalysts. We focus on the enhancement of underlying mechanisms of activity, the interplay between dopant species, and the emerging design principles that govern catalyst performance. Special emphasis is placed on how electronic structure tuning



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via co-doping strategies can generate active sites with optimized hydrogen adsorption energies. While several comprehensive reviews have discussed HER electrocatalysts based on either metallic or non-metallic catalyst strategies, most existing works remain largely material-centric or focus on single-dopant effects, with limited emphasis on comparative or synergistic design principles. In contrast, the present review introduces a unified framework that systematically examines metallic and non-metallic strategies from a mechanism-driven perspective, highlighting how different dopant types modulate electronic structure, defect chemistry, and hydrogen adsorption energetics across diverse material platforms. By explicitly correlating dopant-induced electronic tuning with HER activity and stability, this review goes beyond descriptive classification and provides generalized structure–activity relationships and rational design guidelines for next-generation, Earth-abundant HER electrocatalysts.²⁰

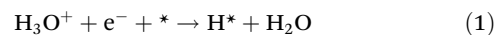
2. Hydrogen evolution reaction mechanism

An imperative electrochemical mechanism called the HER converts protons (H^+) in an aqueous solution into hydrogen gas (H_2). It originates during electrolysis, considered the process of cutting down water into its parts of hydrogen and oxygen by way of an electric current. A typical electrolyzer consists of three fundamental components: (i) an electrolyte, (ii) a cathode, and (iii) an anode. The cathode and anode are generally coated with suitable catalysts to facilitate the electrochemical reactions.²¹ There are several phases to the HER occurring at the cathode, and the electrolyte's state of pH can have an impact on those phases.²² The hydrogen production mechanism within an acidic solution is usually explained by two main phases. At the beginning, H^+ ions near the electrode surface take in electrons for the reduction which forms hydrogen molecules that are adsorbed onto the electrode surface²³ (1). There are two ways in which these adsorbed hydrogen molecules might be eliminated from the electrode surface.

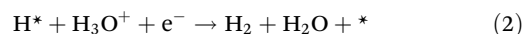
One approach is the Volmer–Tafel mechanism-based mixing of two adsorbed hydrogen atoms (2), which proceeds by their desorption from the electrode surface. The alternative process involves the Volmer–Heyrovsky mechanism-driven response of the adsorbed hydrogen atom with an additional H^+ ion and electron (3). As a result, the latter path usually has an apparent activation energy greater than surface recombination.²⁴

Acidic medium:

Volmer reaction



Heyrovsky reaction



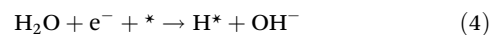
Tafel reaction



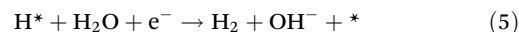
The process of the hydrogen evolution mechanism pathway in an alkaline environment is similar to the acidic environment. However, it moves two or three orders of magnitude less quickly than it moves under acidic circumstances. Furthermore, in alkaline HER the water dissociation phase (4) is where H_{ads} intermediates are formed.²⁵

Alkaline medium:

Volmer reaction



Heyrovsky reaction



Tafel reaction



The development of exceptionally good electrocatalysts requires a knowledge of the HER mechanism. The overall HER mechanism is shown in Fig. 2, where adsorbed hydrogen undergoes a Volmer reaction to generate the intermediate form of H^* . This is followed by a Tafel or Heyrovsky reaction to eliminate hydrogen and release H_2 gas. A standard measure for evaluating the catalyst's efficiency in starting the reaction is the Gibbs free energy of hydrogen adsorption (ΔG_{H^*}), which represents the amount of H^* binding to the electrocatalyst surface.¹⁹ When ΔG_{H^*} is close to zero, it indicates that the adsorption and desorption processes are in equilibrium, which promotes the evolution of hydrogen.²⁶

3. Doping: an electronic structure tuning strategy

Doping, specifically heteroatom doping, is a complex method in materials science that involves deliberately introducing atoms from different elements to fine-tune the electrical structure of materials.²⁷ Many fields depend on this method because it gives researchers unmatched control over the



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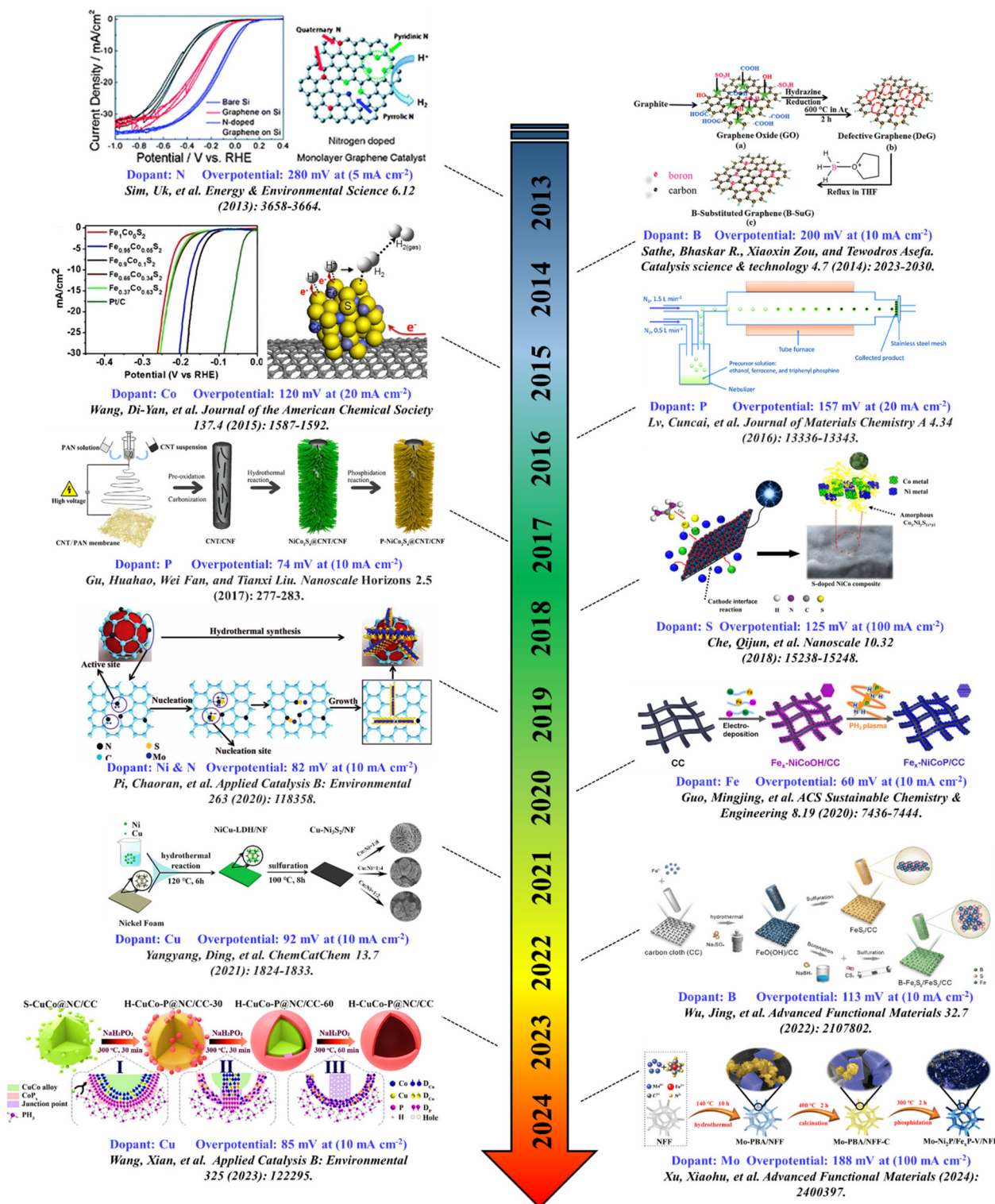


Fig. 1 Schematic representation of historical progress in heteroatom doping for electrocatalytic HER. Reproduced from ref. 45, 62, 74, 121, 151, 168, 174, 198 and 209–212 with permission. Copyright 2013 to 2024.

chemical and physical qualities of materials. Heteroatom doping utilizes the capacity of foreign atoms to modify the electronic band structure of host materials, therefore impacting their conductivity, catalytic activity, and optical properties,

shown in Fig. 3.¹⁴ Researchers may precisely modify the electrical characteristics of materials such as semiconductors, carbon-based materials, and transition metal oxides by deliberately introducing atoms like nitrogen, boron, phosphorus,



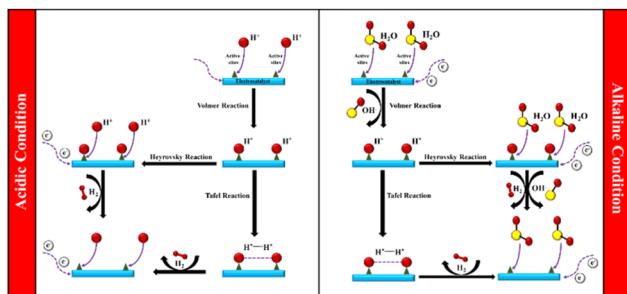


Fig. 2 Acidic and alkaline conditions mechanism for HER.

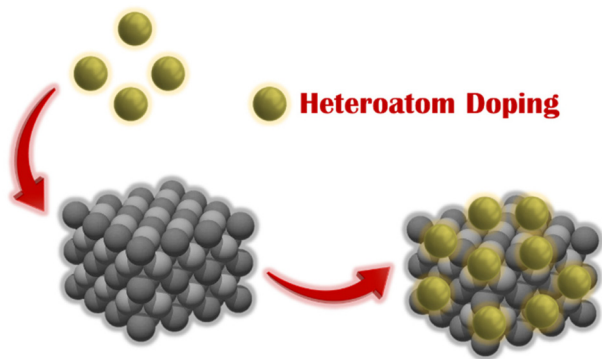


Fig. 3 Schematic illustration of heteroatom doping: incorporation of foreign atoms (yellow spheres) into the host material (gray lattice) to modify its electronic structure and enhance electrocatalytic performance.

sulphur, and others into their lattice structure.²⁸ Customization is essential for a wide range of applications, including semiconductor devices, where doping can control carrier quantities and types (n-type or p-type), and catalysis, where doping improves catalytic efficiency and selectivity.²⁹ Furthermore, heteroatom-doped materials in energy storage systems such as lithium-ion batteries and fuel cells demonstrate higher electrochemical performance due to improved charge transfer kinetics and stability during operation.³⁰ Heteroatom-doped materials are synthesized using sophisticated techniques such as chemical vapor deposition (CVD), sol-gel procedures, and plasma-enhanced processes. These techniques allow for accurate control of the concentration and distribution of the dopants.¹⁵ Characterization techniques such as X-ray photoelectron spectroscopy (XPS), transmission electron microscopy (TEM), and computational modeling methods like density functional theory (DFT) are crucial for comprehending the structural and electrical modifications caused by doping. These methods offer valuable information on the incorporation of heteroatoms into the host lattice, the modification of local bonding arrangements, and the generation of novel electronic states inside the band gap.³¹ A comprehensive comprehension of these fundamental principles is crucial for maximizing the efficiency of material performance and propelling advancements in technological applications. In

addition to fundamental research, the technique of heteroatom doping shows potential for tackling urgent global issues such as environmental cleanup and the development of sustainable energy alternatives.³² Materials that have been infused with heteroatoms demonstrate improved capacities in breaking down pollutants, purifying water, and converting renewable energy. This emphasizes their potential to significantly contribute to environmental sustainability.³³ Challenges in this subject involve the consistent distribution of doping agents in large-scale manufacturing, guaranteeing the durability of doped materials over time, and refining doping techniques for uses.³² Future research endeavours seek to investigate innovative doping techniques, elucidate the underlying principles of doping at the atomic level, and incorporate materials doped with heteroatoms into usable devices and systems. In the field of materials science, heteroatom doping is considered a fundamental aspect that provides numerous possibilities for innovation and the advancement of cutting-edge technologies that will have a significant impact on our future.³⁴

Doping is a widely employed strategy in materials science to tailor the electronic properties of materials. In the context of heteroatom doping, doping involves the intentional introduction of foreign atoms into a material's lattice to modulate its electronic structure.²⁸ This approach enables the tuning of the material's bandgap, Fermi level, and charge carrier density, thereby enhancing its performance in various applications. Heteroatom doping has emerged as a powerful tool for designing materials with tailored electronic properties.^{28,35} By strategically selecting the dopant atoms and their concentrations, researchers can engineer the material's electronic structure to suit specific requirements. For instance, doping can enhance the catalytic activity, optical absorption, or electrical conductivity of materials, making them more suitable for applications in energy storage, optoelectronics, and sensing.³⁶ Theoretical modeling and simulations play a crucial role in understanding the effects of doping on the electronic structure of materials.³⁷ By employing density functional theory (DFT) and other computational methods, researchers can predict the dopant-induced changes in the material's band structure, density of states, and charge distribution. These insights enable the rational design of doped materials with optimized electronic properties, accelerating the discovery of novel materials for various technological applications.³⁸

3.1 Tuning the electronic structure for enhanced electrocatalytic activity

The intrinsic activity of electrocatalysts is heavily influenced by their electronic structure, making it a central focus in catalyst design.³⁹ One effective strategy to boost the water-splitting efficiency of transition metal nanomaterials is heteroatom doping, which modifies their electronic configuration, energy bands, and interaction with reactive species.¹⁴ Incorporating foreign elements has been shown to enhance surface wettability, reduce kinetic barriers, and introduce additional active sites by altering the host material's surface electronic features.



These modifications directly impact the catalyst's ability to bind oxygen-containing intermediates, thereby improving its electrocatalytic behaviour.¹⁶ Such engineering methods can lead to rearrangements in spin and charge distributions, which further amplify catalytic performance.⁴⁰ Additionally, doping can disrupt the original atomic ratios, create localized electric field distortions and shift the electronic density. Overall, tailoring the electronic structure through doping serves as a powerful approach for optimizing Fermi level positioning and surface reaction dynamics, both of which are crucial for efficient overall water splitting.⁴¹

3.2 Modulation of active sites for optimal electrocatalysis

Identifying and understanding the nature of active sites in electrocatalysts is essential for unravelling reaction mechanisms and developing more effective catalytic materials. These active sites are the specific locations on the catalyst surface where electrochemical reactions take place, and their intrinsic reactivity depends on how they interact with key intermediates, such as oxygen- and hydrogen-containing species.⁴² Doping plays a critical role in modifying these sites in two primary ways: (a) by increasing the quantity of accessible active sites, thus raising their overall density and enhancing catalytic output, and (b) by altering the chemical nature of the sites, thereby tuning their affinity and reactivity toward specific intermediates.⁴³ As discussed, the introduction of cations or ionic dopants can fine-tune the properties of active sites, leading to improved reaction kinetics and potentially faster catalytic processes.⁴⁴

3.3 Harnessing synergistic effects through heteroatom doping for enhanced HER

Heteroatom doping has emerged as a highly effective strategy for enhancing HER activity, primarily by modulating the electronic environment and accelerating charge transfer processes.⁴⁵ By introducing dopants with varying electronegativity or valence states, strong synergistic effects can be achieved, leading to optimized electronic structures that enhance intrinsic catalytic activity.⁴⁶ These effects can improve the interaction between the catalyst surface and hydrogen-containing intermediates, reduce energy barriers, and facilitate more efficient electron conduction and interfacial charge transport.⁴⁷ Synergistic modifications also influence surface morphology and increase the density of accessible active sites, which are crucial for improving catalytic kinetics.⁴⁸ Additionally, heteroatom doping can fine-tune the adsorption energies of intermediates involved in the HER, especially the hydrogen adsorption–desorption step, often the rate-determining step in the process. This results in faster reaction rates and lower overpotentials, thus significantly boosting HER efficiency.⁴⁹ However, the benefits of doping depend heavily on maintaining an optimal dopant concentration. Excessive doping may lead to dopant clustering, structural instability, or the formation of secondary phases, which can negatively affect catalytic performance.¹⁶ Therefore, achieving uniform and controlled doping within the solubility limit is vital to preserve

the desired properties. Co-doping strategies involving both cationic and anionic elements can further enhance HER performance through complementary effects on the electronic structure and surface chemistry.⁵⁰ While challenges remain such as understanding dopant distribution, real-time behaviour during catalysis, and correlating dopant levels with catalytic performance, ongoing research continues to refine these approaches. Overall, the rational design of synergistically doped materials offers a promising pathway toward the development of highly efficient and durable HER electrocatalysts.⁴⁸

4. Heteroatom doping strategies for enhanced electrocatalytic HER

Heteroatom doping has emerged as a powerful strategy to enhance the electrocatalytic performance of materials for the HER.⁵¹ Depending on the nature of the dopant, heteroatom doping is broadly categorized into metallic and non-metallic types, shown in Fig. 4.¹⁴ Metallic dopants such as iron (Fe), cobalt (Co), nickel (Ni), molybdenum (Mo), manganese (Mn), copper (Cu), and platinum (Pt) are commonly introduced to improve electrical conductivity, lower the energy barrier for hydrogen adsorption, and facilitate faster charge transfer kinetics.¹⁶ These transition and noble metals serve as active catalytic centers, significantly boosting HER efficiency.⁵² On the other hand, non-metallic dopants like phosphorus (P), nitrogen (N), boron (B), oxygen (O), and sulfur (S) modulate the electronic structure of the catalyst, introduce defect sites, and adjust the binding energy of reaction intermediates. Such modifications enhance intrinsic activity and provide abundant accessible active sites.⁵³ Overall, the deliberate choice of heteroatom dopants – metallic or non-metallic – enables the fine-tuning of catalytic behavior, offering a versatile platform for the design of cost-effective and high-performance electrocatalysts for sustainable hydrogen production. In the following sections, we will systematically explore how each class of dopants, metallic and non-metallic, contributes to HER per-

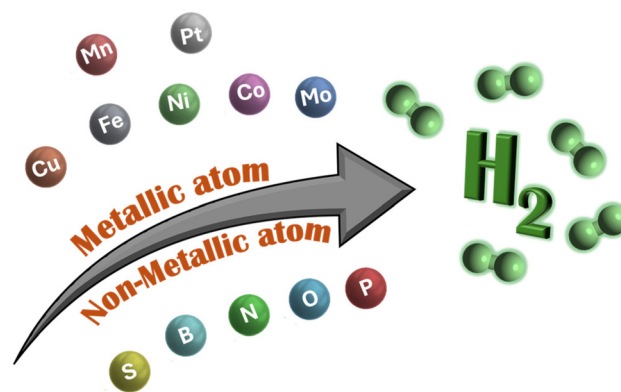


Fig. 4 Doping of metallic and non-metallic atoms to enhance catalytic activity for HER.



formance, supported by recent advancements and material-specific examples.

4.1 Metallic heteroatom doping

The incorporation of metallic atoms into a host matrix is a powerful strategy for engineering advanced catalysts, primarily due to their strong chemical compatibility with the base metal lattice.⁵⁴ This often leads to the formation of highly uniform solid solutions or alloy phases, which can be exploited to finely tune the material's intrinsic properties.⁵⁵ Among various dopants, 3d transition metals, owing to their partially filled or vacant d-orbitals, are particularly effective at boosting catalytic performance. Their influence on catalytic behaviour is often interpreted through the lens of the “volcano plot”, a conceptual framework that correlates catalytic activity with hydrogen adsorption energy (ΔG_{H^*}), especially relevant for the HER, shown in Fig. 5a.⁵⁶ The peak of this curve signifies an optimal balance, where hydrogen binds neither too strongly nor too weakly, enabling the most efficient catalysis. Dopants that shift the catalyst's ΔG_{H^*} value closer to this thermoneutral point can significantly enhance HER efficiency.⁵⁷ However, this enhancement isn't solely dictated by adsorption energy.

As evidenced in comparative studies, shown in Fig. 5b, introducing a range of both noble and non-noble metallic elements can reduce overpotentials markedly from high (blue zone) to low (pink zone), indicating improved catalytic performance.^{56,57} Yet, some dopants achieve remarkable activity even when their ΔG_{H^*} values deviate from the volcano peak, suggesting additional contributing factors. This implies that the catalytic output is also shaped by the local atomic environment and the electronic structure of the host material. As shown in Table 1, metal dopants mainly improve how hydrogen binds to the catalyst, while non-metal dopants help by changing the surface environment and active sites. Dopant-induced changes in coordination geometry and electron distribution can alter active site behaviour beyond simple thermodynamic predictions. Thus, in understanding the role played by heterometallic dopants in non-noble metal catalysts for HER, it's essential to consider not only adsorption energetics but also the complex interplay of structural and electronic modifications introduced by doping. Selected examples highlighting these effects will be explored in the following sections.

4.1.1 Fe-doped electrocatalysts for HER. Fe atoms' versatility and abundance of d-band electrons enable them to suc-

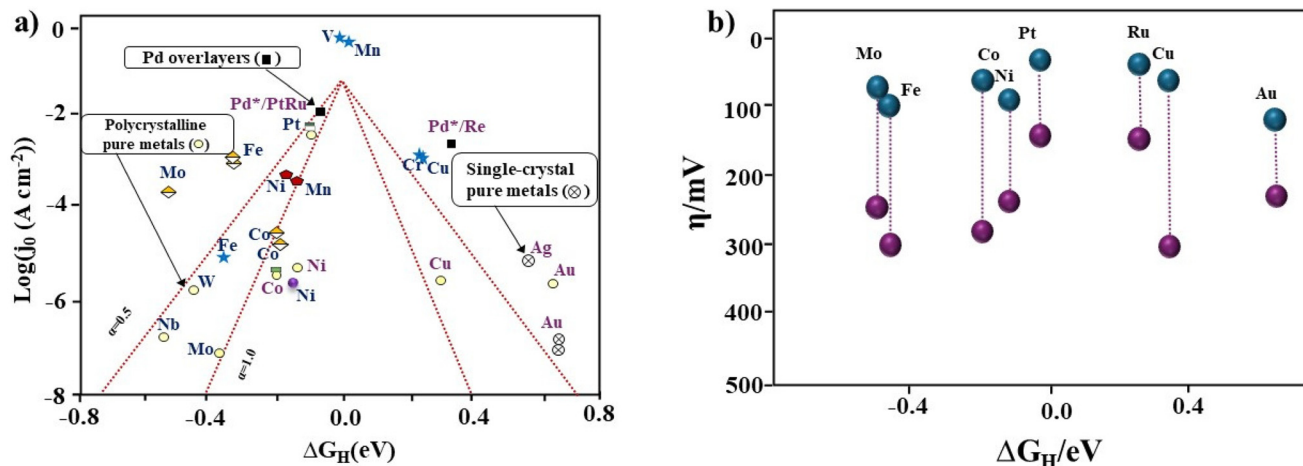


Fig. 5 (a) Correlation between HER activity and hydrogen ΔG_{H^*} for various metallic systems, including heteroatom-doped non-noble metal catalysts, is represented. (b) The overpotential of non-noble metal catalysts doped with metallic heteroatoms is directly influenced by the ΔG_{H^*} of the catalytic sites, which governs their hydrogen evolution efficiency.

Table 1 Key differences between metallic and non-metallic dopants in hydrogen evolution reaction catalysts

Key aspect	Metallic dopants	Non-metallic dopants
Electronic modulation	Modify the electronic structure of metal centers, often through d-band tuning and metal-metal interactions, which influences catalytic behaviour	Redistribute charge density around host atoms, producing polarized sites and altering surface electronic environments
Effect on ΔG_{H^*}	Regulate hydrogen adsorption free energy more directly <i>via</i> metal-hydrogen interactions	Influence ΔG_{H^*} indirectly by altering the local electronic structure of neighbouring atoms
Nature of active sites	Generate metal-centered or bimetallic active sites responsible for HER	Produce heteroatom-based or metal-heteroatom (M-X) sites that participate in hydrogen adsorption
Structural influence	Induce lattice distortion, defect formation or phase evolution within the catalyst structure	Promote defect generation, edge exposure and surface functionalization
Role in HER performance	Primarily enhance intrinsic catalytic activity and charge-transfer kinetics	Improve surface reactivity and contribute to stabilization of reaction intermediates



cessfully control the electronic structure of host catalysts.¹⁶ Due to their comparable atomic radii and electrical structures, Co and Ni are examples of substrate metal atoms that have been the focus of most research on Fe-doped catalysts until now. An appropriate amount of iron salt is usually added as an intermediate during synthesis to accomplish Fe doping. One of the main processes resulting from Fe doping is that Fe alters electronic structure and lowers ΔG_{H^+} .^{58–60}

Xiaosong Guo and co-workers reported an alternative route for the manufacture of Fe–Co Prussian Blue Analogue (PBA) nanoclusters, responsible for overall increased catalytic performance, employing room-temperature layer-structured Co(OH)₂ nanoparticles as sacrificial templates, shown in (Fig. 6a). To achieve extremely uniform Fe doping at the atomic level, the multilayered Co(OH)₂ nanocrystal matrix first intercalated and evenly anchored [Fe(CN)₆]^{3–} anions. Stable Fe–Co PBA catalysts were then formed when the embedded [Fe(CN)₆]^{3–} anions interacted with neighbouring Co(OH)₂ host layers. Crucially, these Fe–Co PBA catalysts embedded themselves in the Co(OH)₂ substrate layers, hence inhibiting Fe–Co PBA nanoparticle aggregation and hyperproliferation. XPS showed that Fe inclusion caused negative shifts in the Co 2p and P 2p peaks, indicating altered electronic states. These modifications strengthened Fe–Co interactions and boosted HER activity. The Fe–Co_xP catalyst exhibited a significant increase in HER catalytic activity. This offers a straightforward and efficient way to achieve heteroatom doping.⁶¹ Mingjing Guo and co-workers developed an electrode placement and PH₃ plasma treatment that was used to effectively construct Fe-doped Ni–Co phosphide nanoplates that incorporated a hierarchical nanostructure on a carbon cloth (Fe_x–NiCoP), shown in Fig. 6b. The electrocatalytic activity was further enhanced by the planar flaws in the nanocrystals caused by Fe doping, which encouraged the creation of active sites. Ultimately, the electrocatalytic efficacy of the catalyst may also be influenced by potential electronic interactions between metal cations following Fe doping with the right amount of content. After phosphorization, XPS results showed the creation of partly charged Ni^{δ+}, Co^{δ+}, Fe^{δ+}, and P^{δ–} species, indicating strong electronic interactions. These characteristics allowed proton adsorption and improved electrocatalytic performance. DFT calculations showed that slight Fe doping improved HER activity by promoting H₂O adsorption on active sites. Fe₁–NiCoP with Fe doping had the highest HER performance, with stability lasting 25 h and an η_{10} of just 60 mV.⁶² Chun Tang and co-workers reported that a Fe-doped CoP nanoarray on Ti foil (Fe–CoP/Ti) was a strong monolithic HER catalyst with higher activity than a CoP nanoarray. Because of their distinct valence, ionic radius, and ion mass, Co²⁺ adsorbed on Ti foil at a higher pace than Fe³⁺, which may explain why the Fe/Co atomic proportions in the solutions and final products differed. In a two-electrode electrolyzer, an HER overpotential with 78 mV for 10 mA cm^{–2} necessitated a voltage in the cell of 1.60 V for water-splitting current. With an activating energy as low as 39.6 kJ mol^{–1}, it yielded a hydrogen production rate of 6.06 L min^{–1} g^{–1} for 1 wt% NaBH₄ under ambient con-

ditions.⁶³ Tadele Hunde Wondimu and co-workers reported Fe-doped WO₃ deposited directly over reduced graphene oxide in the form of a nanoplate-like structure, and reduction was then accomplished using PH₃ produced *in situ* using sodium hypophosphite. Fe-doped tungsten trioxide nanoplates (Fe–WO₃) were converted into Fe-doped tungsten oxide phosphate (Fe–WO_xP) by PH₃, which functioned as a phosphorus precursor and reducing agent. Fe–WO_xP on reduced graphene oxide (Fe–WO_xP/rGO) was the product of this. The numerous oxygen vacancies within the nanoplate-like framework of Fe–WO_xP and the rGO nanosheets worked in concert to increase the catalytic ability and conducting capacity of the synthesized material. In 0.5 M H₂SO₄, Fe–WO_xP/rGO demonstrated remarkable electrocatalytic activity despite a low overpotential of 54 mV, which was adequate to attain a current density of 10 mA cm^{–2} and stable for more than 24 h.⁶⁴ Junqing Yan and co-workers developed Fe-doped NiS₂ (Fe–NiS₂) nanosheets that exhibited strong activity and repeatability in the electrochemical process. Fe³⁺ was added to the lattice to increase the NiS₂ semiconductor's conductivity. Fe-doped NiS₂ nanosheets (Fe–NiS₂) were produced by sulfidating NiFeO_x/α-Ni(OH)₂ nanosheets during the production process. Interestingly, on the surface of the Ni(OH)₂ nanosheet, the previously dispersed NiFe₂O₄ nanoparticles changed into sulfide and integrated into the crystal structure of the ensuing NiS₂ sample. DFT calculations revealed that the electrocatalytic HER transition state could lower the activation energy of H₂ production. After sulfuration, the material preserves its original elemental framework with the successful incorporation of sulfur, as confirmed by XPS analysis. Enhanced HER capabilities in the resulting Fe–NiS₂ nanosheet sample resulted from adding surface Fe sites, which boosted electrocatalytic performance. These attributes included a tiny overpotential of 121 mV at 10 mA cm^{–2} and a low Tafel slope of 37 mV dec^{–1}.⁶⁵ Cheng Wan and co-workers first reported a unique amine–metal oxide synthesis technique: molybdenum carbide nanoparticles with the Fe₂N composition could be doped with varying concentrations of iron (0–8 wt%). A graphitic carbon framework was formed by the Fe-doped catalyst during the β-Mo₂C synthesis process. The Fe dopant serves as an effective catalyst that facilitates the formation of graphitic carbon during the β-Mo₂C synthesis process. There was no discernible difference in the Fe oxidation states between 0 and 2+ in the Fe-doped β-Mo₂C. The valence bands (VB) were altered by the partly oxidized Fe, leading to wider VB. Tests using the HER indicated that Fe-doped β-Mo₂C was a more active electrocatalyst than pure β-Mo₂C made from different sources of molybdenum. The wider valence bands of β-Mo₂C, more carbon-based graphitic supports, and the highly active Fe₂(MoO₄)₃ product on the outermost layer following Fe doping were all thought to be responsible for this increased activity, shown in Fig. 6c.⁶⁶

Feng Li and co-workers have reported that different Fe doping levels for the optimized porous cobalt phosphide polyhedron indicated substantially better HER performance. It had lower onset overpotentials, higher current densities, lower Tafel slopes, and superior electrochemical stability throughout



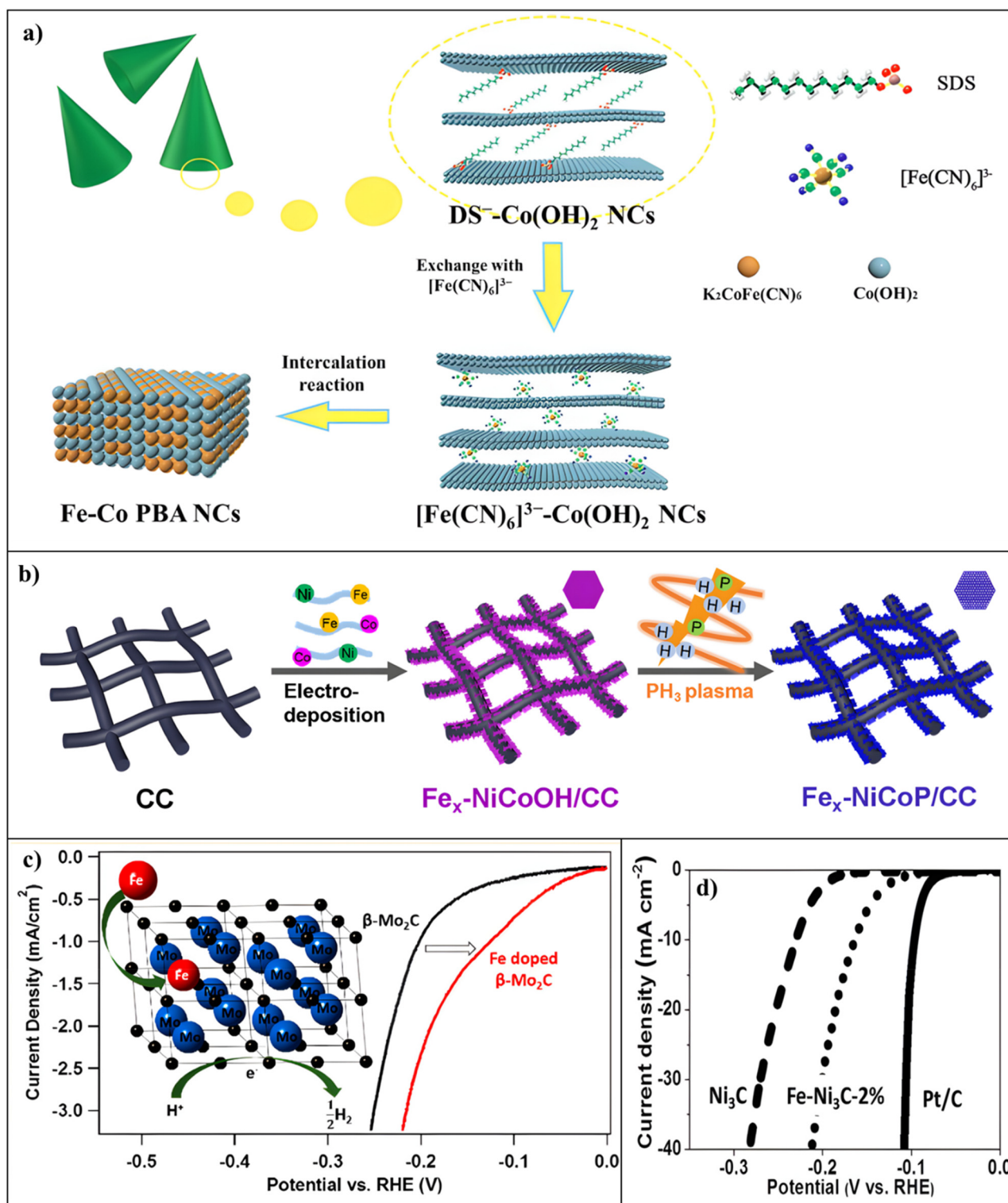


Fig. 6 (a) Schematic illustrating the mechanism by which Fe-Co PBA NCs emerges from $\text{Co}(\text{OH})_2$ NCs. Reproduced from ref. 61 with permission. Copyright 2018, ACS. (b) A diagrammatic illustration of the Fe_x -NiCoP/CC synthesis method. Reproduced from ref. 62 with permission. Copyright 2020, ACS. (c) Polarisation curves for Fe-doped β - Mo_2C and pure β - Mo_2C . Reproduced from ref. 66 with permission. Copyright 2015, ACS. (d) Sample polarisation curves for Ni_3C , Fe-Ni $_3\text{C}$ -2%, and Pt/C in 0.5 M H_2SO_4 . Reproduced from ref. 70 with permission. Copyright 2017, Wiley.



the HER. By using Co and Fe-containing MOFs in a one-step phosphorization procedure at a low temperature, porous polyhedrons containing CoP doped with Fe were developed. The $\text{Co}_{0.68}\text{Fe}_{0.32}\text{P}$ precursor sample, according to morphological examination, had a structure made up of uniformly dispersed MIL-88B related to ZIF-67 polyhedra each with an average size of 2 μm . The energy-dispersive spectroscopic spectra of Co and Fe were used to validate their elemental makeup. $\text{Co}_{0.68}\text{Fe}_{0.32}\text{P}$ outperformed previous non-noble-based electrocatalysts described in the literature in terms of long-term stability, and significantly enhanced electrocatalytic activity for the HER with overpotentials as low as 116 mV in 10 mA cm^{-2} .⁶⁷ Pengyan Wang and co-workers have reported that $\text{Ni}_{1.85}\text{Fe}_{0.15}\text{P}$ nanosheet arrays over nickel foam electrodes (NSAs/NF) demonstrate exceptional and very stable electrocatalytic activity for the HER. Fe-doping caused a small shift in the diffraction peaks to greater angles, indicating that Fe ions had been substituted within the Ni_2P lattices. Because the atomic sizes of Ni and Fe are comparable, the crystalline structure of Ni_2P does not change when Ni is replaced with Fe. The resulting $\text{Ni}_{1.85}\text{Fe}_{0.15}\text{P}$ nanosheet arrays over a (NSAs/NF) electrode showed extraordinary electrocatalytic activity in the HER and they required just 106 mV overpotential to reach a current density of 10 mA cm^{-2} and to remain stable for 20 h.⁶⁸ Can Lin and co-workers have reported that Fe- Co_xP NPs were grown in a single step by adding iron while a metal-organic framework (ZIF-67) was synthesized by an *in situ* method and then treated with phosphate. An insignificant amount of iron phosphide was formed which helped with charge redistribution and improved the electrocatalyst's catalytic activity in concert. The performance of CoP's HER was enhanced by the addition of Fe. This enhancement can be ascribed to the difference in the electronegativities between the Fe and Co, which promotes the faster rate of H_2O breakdown and increases the adsorption and bonding capacity to the intermediate H produced during the HER. XPS revealed successful Fe doping in CoP, with dominant Co-P bonding and minor Fe-P species, whereas surface oxidation introduced M-O bonds. The small amount of iron phosphide caused charge redistribution, which improved electrocatalytic performance. The optimized $\text{Fe}_{0.27}\text{Co}_{0.73}\text{P}$ exhibited a very low overpotential of 186 mV at a current density of 10 mA cm^{-2} , and it remained stable for an amazing 40 h.⁶⁹ Hasosen Fan and co-workers

have developed a remarkable HER performance using Ni_3C nanosheets that operate doped with 2.0 at% Fe nanodots. The electrical characteristics and surface properties of Ni_3C were optimized by the addition of Fe which resulted in increased catalytic efficiency during the generation of hydrogen. In particular, in a KOH solution, the Fe-doped Ni_3C nanosheets demonstrated exceptional electrocatalytic characteristics for HER, including a modest Tafel slope of 41.3 mV dec^{-1} and a low overpotential of 292 mV, shown in Fig. 6d.⁷⁰ The Fe-doped HER electrocatalysts are summarised in Table 2.

4.1.2 Co-doped electrocatalysts for HER. Cobalt doping is a tactical technique used in electrocatalysis particularly to improve catalyst performance in the HER. Small quantities of cobalt are added to the catalyst lattices, which can effectively change the electronic structure. When cobalt is doped into catalyst materials it encourages the adhesion and breakdown of water molecules, which increases the overall reaction efficiency. And because of its modest H-bonding energy cobalt is thought to be a possible catalyst for the HER. Cobalt doping has several benefits, including low cost, stability in electrochemical settings, and advantageous electrical characteristics.⁷¹⁻⁷³

As reported by Di Yan Wang and co-workers, Co-doped nanosheets of iron pyrite FeS_2 were synthesized and closely bonded with nanotubes of carbon ($\text{Fe}_{1-x}\text{Co}_x\text{S}_2/\text{CNT}$ hybrid catalysts). To crystallize the $\text{Fe}_{1-x}\text{Co}_x\text{S}_2$ catalysts and decrease the CNT a high-temperature solvothermal technique was employed in conjunction with the low-temperature fluid phase reaction during the synthesis. Comparing the $\text{FeS}_2\text{-CNT}$ and $\text{Fe}_{0.9}\text{Co}_{0.1}\text{S}_2\text{-CNT}$ surfaces with the Pt (111) surface, the former showed similar molecules of hydrogen adsorption energy (EH_2) and just slightly greater hydrogen adsorption energy (EH). The findings demonstrated a low overpotential of about 120 mV at 20 mA cm^{-2} and long-term durability that exceeded 40 h of continuous HER function. DFT results showed that addition of Co to FeS_2 catalysts lowered the energy barrier for hydrogen atom adsorption in transition state₁ (TS_1).⁷⁴ Guoyuan and co-workers first reported a unique catalyst Co-WSe₂/MWNTs that was effectively synthesized as a heterostructure for the electrocatalytic HER process, shown in Fig. 7a. By using the MWNTs for a very effective topological template, curved Co-WSe₂ nanosheets were easier to nucleate and develop, resulting in an abundance of readily available

Table 2 Fe-doped electrocatalysts for HER

S. no	Catalyst	Electrolyte	Overpotential (η) at 10 mA cm^{-2}	Tafel slope (mV dec^{-1})	C_{dl} (mF cm^{-2})	Stability (hours)	Ref.
1	Fe- Co_xP	0.5 M H_2SO_4	127	55	—	4 h	61
2	$\text{Fe}_x\text{-NiCoP}$	0.1 M KOH	60	51	71.40	25 h	62
3	Fe- CoP/Ti	0.1 M KOH	78	75	—	20 h	63
4	Fe- $\text{WO}_x\text{P/rGO}$	0.5 M H_2SO_4	54	41	34.60	24 h	64
5	Fe- NiS_2	0.5 M H_2SO_4	121	37	—	15 h	65
6	Fe-doped $\beta\text{-Mo}_2\text{C}$	0.1 M HClO_4	240	—	—	9 h	66
7	$\text{Co}_{0.68}\text{Fe}_{0.32}\text{P}$	0.1 M KOH	116	64	—	1 h	67
8	$\text{Ni}_{1.85}\text{Fe}_{0.15}\text{P}$	1.0 M KOH	106	47	6.04	20 h	68
9	$\text{Fe}_{0.27}\text{Co}_{0.73}\text{P}$	1.0 M KOH	186	59	1.02	40 h	69
10	Fe- $\text{Ni}_3\text{C-2\%}$	1.0 M KOH	292	41	—	10 h	70



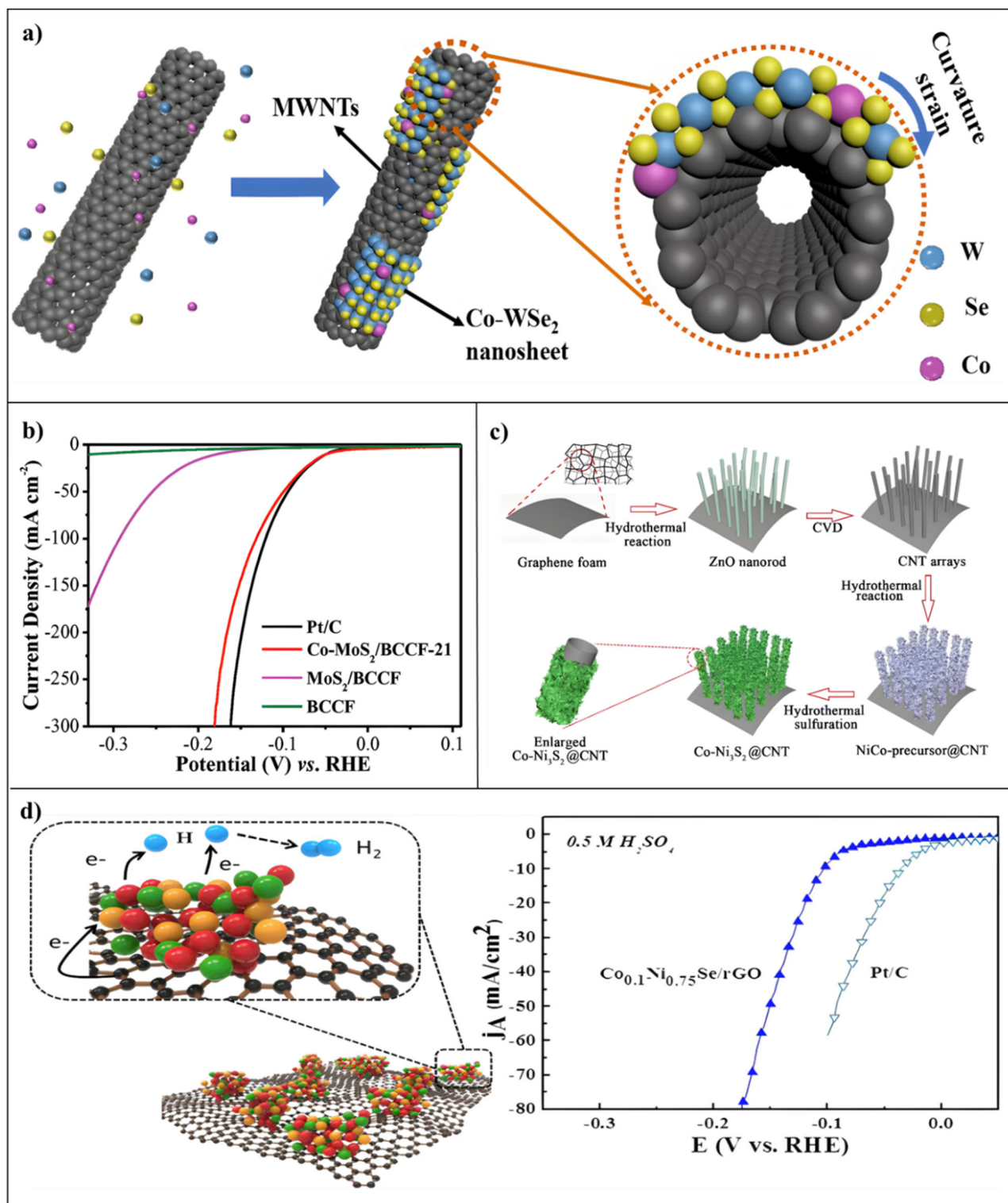


Fig. 7 (a) A schematic representation of the heterostructure formed by Co-doped WSe₂/MWNT synthesis. Reproduced from ref. 75 with permission. Copyright 2018, RSC. (b) The HER polarisation curves for Co-MoS₂/BCCF-21. Reproduced from ref. 76 with permission. Copyright 2018, Wiley. (c) A demonstration of the Co-Ni₃S₂@CNTs/GNF synthesis process. Reproduced from ref. 79 with permission. Copyright 2013, RSC. (d) LSV curves for Co_{0.1}Ni_{0.75}Se/rGO. Reproduced from ref. 81 with permission. Copyright 2018, ACS.



active sites. Cobalt atoms increased the inherent activity of these active sites in TMDs by bringing the ΔG_{H} closer to 0 eV. Because of MWNTs' remarkable conductivity and the close contact between Co-WSe₂ nanosheets along with MWNT, electrons may move quickly across the catalyst. The resultant Co-WSe₂/MWNTs showed an overpotential of 174 mV at 10 mA cm⁻², promising electrocatalytic capability.⁷⁵ Bora Seo and co-workers have reported that the successful addition of cobalt to MoS₂ by covalent doping was demonstrated by the notable enhancement in HER activity. This alteration successfully affected the electronic configuration of MoS₂ in addition to causing dynamic alterations in the atomic structures of the catalyst under *operando* circumstances. XPS showed mixed Mo⁴⁺/Mo⁵⁺/Mo⁶⁺ states and increasing surface oxidation with decreasing MoS₂ layer number, indicating the development of distorted 1T-MoS₂ with more exposed edge sites. The transition from S²⁻ to unsaturated S₂²⁻ ligands, along with increased oxidation, led to more active sites and better HER activity. DFT study showed the Mo-adjusted electron density lowered the free energy of hydrogen adsorption, which eventually improved HER performance. Interestingly, MoS₂ doped with covalent coupling cobalt had a low overpotential of around 48 mV at 10 mA cm⁻², shown in Fig. 7b.⁷⁶ Huanlei Lin and co-workers developed the simple cobalt doping approach used to improve the electrocatalytic HER on Mo₂C nanowires using Co-modified MoO_x-amine precursors. When Co-doping is effective inside the Mo₂C crystal structure the electron density increases near the Fermi level, weakening the Mo-H bond and promoting HER kinetics. Co is already well known for maximizing the catalytic activity of active components in electrocatalysis. Its effectiveness is attributed to surface composition and customized electrical characteristics which are essential for catalytic turnover. The Co-Mo₂C nanowires showed a modest overpotential of 140 mV at 10 mA cm⁻² as anticipated.⁷⁷ Isolda Roger and co-workers reported the direct synthesis of a ternary chalcogenide with its constituents Co₂Mo₉S₂₆ using fluorine-doped tin oxide (FTO) electrode that produced a compound with good HER activity. The direct hydrothermal depositing of metal chalcogenides onto thin oxide electrodes has not yet been reported despite the simplicity and scalability of these procedures. This development might simplify the field by removing the requirement for distinct stages for catalyst deposition and hydrothermal synthesis. XPS revealed the presence of Mo, Co, and S on the film surface, with Mo primarily in the Mo⁴⁺ state and minor Mo⁵⁺/Mo⁶⁺ species from surface oxides. Co had mixed Co²⁺/Co³⁺ valence states, while sulfur mostly occurred as S²⁻, which was compatible with MoS₂-based materials. A high-resolution spectrum in the Co 2p region confirmed the presence of cobalt on the film surface exhibiting a dual state of Co²⁺ and Co³⁺ that supports increased HER activity. In acidic conditions, these films demonstrated impressive HER activity, attaining current densities of 10 mA cm⁻² at a moderate overpotential of 260 mV.⁷⁸ Feifei Wang and co-workers reported robust electrical conductivity demonstrated by a CNT/GNF hybrid with a three-dimensional in nature conductive network that included

Co-doped Ni₃S₂ nanoparticles, produced using a simple hydrothermal technique, shown in Fig. 7c. During the redox process, the tectorum-like Co-Ni₃S₂ nanosheet geometry improved both ion and electron transport. When cobalt doping was added the electrochemical characteristics were greatly enhanced and, in comparison with other materials, the theoretical specific capacity was exceptional. In alkaline solution, the Co-Ni₃S₂@CNT/GNF electrode had excellent HER activity, attaining the lowest overpotential of 155 mV at 10 mA cm⁻².⁷⁹ Xiaoping Dai and co-workers have reported the cobalt doping of molybdenum disulfide, carried out using a deposition and precipitation technique to guarantee that it would not interfere with industrial operations. A thorough investigation was conducted into the effect of Co doping on molybdenum disulfides for HER performance. CoMoS-2 resembled MoS₃ with many bridging disulfides (S₂²⁻), while CoMoS-2-C and MoS-C were closer to MoS₂ due to Co substitution at MoS₂ edges and enhanced crystallinity. The XPS spectra confirmed dominant, well-dispersed CoMoS phases on carbon, with mixed Mo oxidation states, S₂²⁻/S²⁻ ligands, and little Co₉S₈ production. DFT calculations revealed a trend in HER activity in doped catalysts. The optimized Co-doped MoS₂ catalyst exhibited a low onset potential, favorable Tafel slope, and a high exchange current density, all of which significantly improved HER performance. The generation of a Co-Mo-S phase was responsible for this enhancement, as was the efficient reduction of charge-transfer impedance, the promotion of structural and electrical modulations among MoS₂ and Co and the optimization of active surface area. With its low starting potential of 90 mV, modest Tafel slope of 50 mV dec⁻¹ and high exchange current density of 0.03 mA cm⁻², the optimised CoMoS-2-C catalyst exhibited remarkable HER performance.⁸⁰ Unique cobalt-doped Ni_{0.85}Se chalcogenides (Co_xNi_{0.85-x}Se) have been reported by Wenjun Zhao and co-workers as stable and extremely effective catalysts for the hydrogen evolution process during the electrolysis of water. The Co_xNi_{0.85-x}Se catalysts had unique physical characteristics when compared to pure Ni_{0.85}Se, leading to diverse catalytic outcomes for HER. In comparison to unsupported Co_{0.1}Ni_{0.75}Se catalysts, experimental data showed that Co_{0.1}Ni_{0.75}Se/rGO displayed a smaller Tafel slope and a lower overpotential. The incorporation of rGO was associated with this improvement because it effectively stopped Co_{0.1}Ni_{0.75}Se from combining, exposing more active sites and enhancing conductivity and ECSA. By reducing the HER overpotential by 103 mV at a current density of 10 mA cm⁻², the Co_{0.1}Ni_{0.75}Se/rGO catalyst produced a lower Tafel slope of 43 mV dec⁻¹, shown in Fig. 7d.⁸¹ The electrochemical activity of the (Cu_{0.83}Co_{0.17})₃P:30S catalyst for the HER process has been reported by Prasad Prakash Patel and co-workers. Like noble metal catalyst systems, the simultaneous incorporation of S and Co into the Cu₃P lattice offers a unique chance to modify its electronic configuration and physical, electrical, and electro-catalytic capabilities. When compared to Pt/C, the (Cu_{0.83}Co_{0.17})₃P:30S system showed equivalent or better-applied bias photon-to-current efficiency during evaluation as



a HER catalyst. Theoretical and experimental results demonstrated that $(\text{Cu}_{0.83}\text{Co}_{0.17})_3\text{P}:\text{30S}$ nanoparticles displayed Pt-like electrochemical performance across all pH conditions. DFT and XPS confirmed that the insertion of Co and S into the Cu_3P lattice affected the electrical structure. The development of a single-phase hexagonal structure resembling Cu_3P was confirmed by structural study. Importantly, at 10 mA cm^{-2} , the produced catalyst showed a low overpotential of 46 mV.⁸² Cobalt doping in $\text{MoS}_2/\text{NiS}_2$ nanosheet array catalysts has been reported by Yunhua Zheng and co-workers to improve the hydrogen evolution process. By combining Co- MoS_2 nanosheets in order with NiS_2 nanosheets, a simple three-step hydrothermal technique was used to produce the Co-doped $\text{MoS}_2/\text{NiS}_2$ nanosheet arrangement *in situ* on a carbon cloth substrate (Co- $\text{MoS}_2/\text{NiS}_2/\text{CC}$). The Co- $\text{MoS}_2/\text{NiS}_2/\text{CC}$ structure that resulted from the addition of Co atoms and multiple heterostructures provided high conductivity and a large number of accessible active sites, which promoted a synergistic effect that improved the electrocatalytic capabilities for HER. Additionally, efforts have been directed at enhancing MoS_2 electrocatalysis using morphological management, heterostructure development, heteroatomic doping, and the use of conductive substrates to raise conductive properties and active site accessibility. Interestingly, the optimized Co- $\text{MoS}_2/\text{NiS}_2/\text{CC}$ electrocatalyst showed outstanding electrocatalytic activity simultaneously in 0.1 M KOH, with a low overpotential of 92 mV and 122 mV to generate current densities of 10 and 50 mA cm^{-2} in 0.5 M H_2SO_4 . In addition, the catalyst exhibited remarkable long-term stability with a Tafel slope of just 47 mV dec^{-1} .⁸³ The Co-doped HER electrocatalysts are summarised in Table 3.

4.1.3 Ni-doped electrocatalysts for HER. Referring to the widely known volcano-type curve for the HER, nickel (Ni) metal is located close to the peak, indicating that it can successfully modify the ΔG_{H^*} for the benefit of HER catalysis. Observations have shown that Ni^{2+} ions could break the H-OH connection, which accelerates the breakdown of water molecules into protons. As in the cases of iron (Fe) and cobalt (Co) doping the primary mechanism underlying the increased activity of Ni-doped electrocatalysts is the modification of the electronic configuration that occurs following the addition of Ni atoms. With strong metal-support contacts Ni-doped catalysts exhibit increased stability and reduce catalytic degra-

dation over prolonged electrochemical cycling. By altering surface charge distribution and promoting proton transfer activities at the catalyst-electrolyte interface, nickel doping also affects electrolyte relationships. Moreover, nickel addition modifies TMDs' electrical structure, promoting effective electron transport during the HER.^{91,94,95}

Xin-Yao Yu and co-workers have reported Ni-Co- MoS_2 nanoboxes showing increased electrocatalytic activity for hydrogen evolution. Uniform Ni-Co Prussian blue analogue nanocubes were produced using a modified precipitation technique. Following that, nickel and cobalt were doped into MoS_2 nanosheets. Specific Ni-Co- MoS_2 nanoboxes made from very thin nanosheets were the final product of these methods. The hollow structure and large surface area of ultrathin MoS_2 nanosheets promote electrochemical responses; edge-terminated structures boost electroactive sites; defect-rich MoS_2 nanosheets reveal more active edges; and doping of transition metals optimises electrical structures and expands interlayers; these are some of the reasons for the enhanced HER activity of the Ni-Co- MoS_2 nanoboxes. Considering an initial potential of 125 mV and a modest Tafel slope of 51 mV dec^{-1} for HER these properties together improved the electrochemical activity of Ni-Co- MoS_2 nanoboxes.⁸⁴ Ni-incorporated amorphous FeP nanoparticle porous TiN nanowires and graphitic carbon fibers (Ni-FeP/TiN/CC) were synthesized in a hierarchical structure by Xiang Peng and co-workers using a step-by-step method. To modify the electrical structure of the FeP/TiN/CC composite to produce an amorphous surface, they used plasma implantation of Ni ions. The HER efficiency was greatly increased by this simultaneous doping and amorphization technique. An active amorphous surface, a conducting nanowire scaffold, and the complementary actions of Ni and Fe atoms in the nanoparticles were responsible for the catalyst's remarkable efficiency. High densities of active sites were exposed by these characteristics, which also enhanced charge transfer effectiveness and inhibited catalyst movement and formation. For an overpotential of 75 mV and a cathodic current density of 10 mA cm^{-2} the Ni-FeP/TiN/CC catalyst exhibited outstanding HER performance, shown in Fig. 8a.⁸⁵ Direct fabrication of Ni-doped Mo_2C nanowires on Ni Foam was reported by Kun Xiong and co-workers using a hydrothermal reaction together with a carburization procedure. The rationale for the increased catalytic activity of Ni $\text{Mo}_2\text{C}/\text{NF}$ was the higher intrinsic

Table 3 Co-doped electrocatalysts for HER

S. no	Catalyst	Electrolyte	Overpotential (η) at 10 mA cm^{-2}	Tafel slope (mV dec^{-1})	C_{dl} (mF cm^{-2})	Stability (hours)	Ref.
1	$\text{Fe}_{0.9}\text{Co}_{0.1}\text{S}_2\text{-CNT}$	0.5 M H_2SO_4	120 at (20 mA cm^{-2})	46	—	40 h	74
2	Co- WSe_2/MWNT	1.0 M KOH	174	37	1.9	3 h	75
3	Co- $\text{MoS}_2/\text{BCCF2}$	1.0 M KOH	48	52	—	12 h	76
4	Co- Mo_2C nanowires	0.5 M H_2SO_4	140	39	7.51	24 h	77
5	$\text{Co}_2\text{Mo}_6\text{S}_{26}$	0.5 M H_2SO_4	260	64	—	4 h	78
6	Co- $\text{Ni}_3\text{S}_2@\text{CNT}/\text{GNF}$	6.0 M KOH	115	138	—	20 h	79
7	Co $\text{MoS}_2\text{-C}$	0.5 M H_2SO_4	135	50	—	3 h	80
8	$\text{Co}_{0.1}\text{Ni}_{0.75}\text{Se}/\text{rGO}$	0.5 M H_2SO_4	103	43	1.71	30 h	81
9	$(\text{Cu}_{0.83}\text{Co}_{0.17})_3$	0.5 M H_2SO_4	46	32	—	24 h	82
10	Co- $\text{MoS}_2/\text{NiS}_2/\text{CC}$	0.5 M H_2SO_4	92	47	—	40 h	83



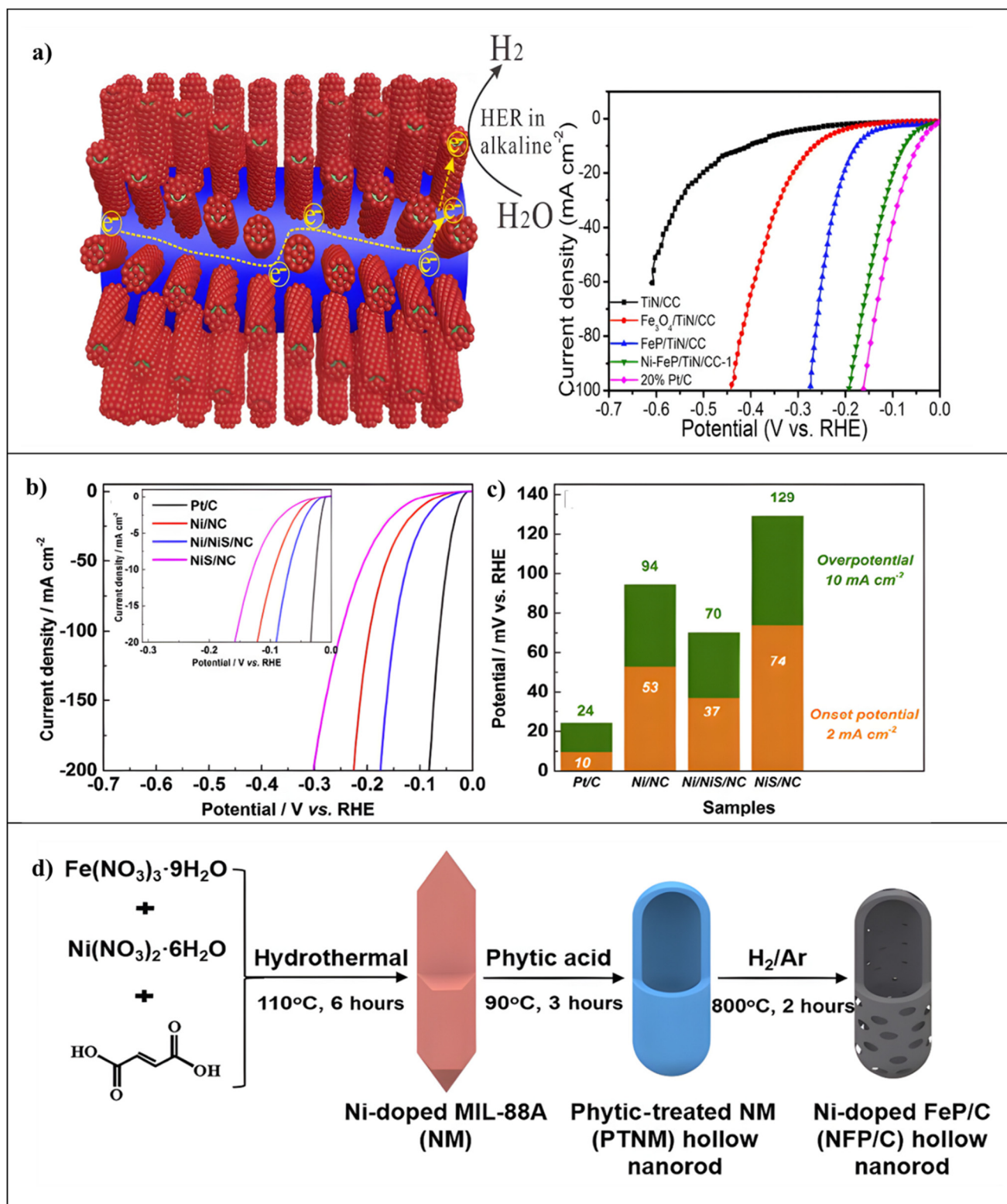


Fig. 8 (a) A diagrammatic illustration of the hierarchical Ni-FeP/TiN/CC electrocatalyst preparation process and curves of polarisation. Reproduced from ref. 85 with permission. Copyright 2018, Elsevier. (b) Ni/NC, Ni/NiS/NC, NiS/NC, and commercial Pt/C catalysts LSV curves in 1 M KOH at a scan rate of 5 mV s⁻¹; (c) the catalysts spectrum of onset and overpotential reproduced from ref. 89 with permission. Copyright 2018, Elsevier. (d) Diagrammatic representation of the Ni-doped FeP/C hollow nanorod production. Reproduced from ref. 93 with permission. Copyright 2019, Science.



sic activity that results from the synergistic interactions between Ni and Mo₂C. This was made possible by the 1D structure's high aspect ratio and large surface area, which exposed multiple Ni-doped Mo₂C sites. The computations using DFT showed that adding Ni to Mo₂C modified the charge distribution of the catalyst, resulting in a synergistic effect that lowered the hydrogen binding energy. In addition to a low starting overpotential of 21 mV, electrochemical studies showed that the NiMo₂C/NF catalyst that was developed had good catalytic effectiveness for producing hydrogen. XPS confirmed Mo₂C development with negligible surface oxides and no metallic Mo, which aligned with XRD data. The presence of Ni caused Mo 3d peak shifts and partial phase separation. Ni existed as both metallic and substituted species inside the Mo₂C lattice. Moreover, only 150 mV overpotential was needed to attain a 100 mA cm⁻² cathodic current density.⁸⁶ A 3D hierarchical catalyst made from Ni₃FeN for water splitting has been reported by Zhihe Liu and co-workers. To improve electronic conductivity and produce 3D conductive networks, carbon cloth was used as the scaffold. Ni₃FeN in the composite was produced using electrodeposition, which provides an effective process appropriate for large-scale manufacturing. A study into the bimetallic Ni₃FeN's development mechanism showed that Ni₃Fe functioned as an intermediary throughout the nitridation process, affecting the final Ni₃FeN nano framework shape. A suggested mechanism highlights how both cationic atoms Ni and Fe work together to improve electrocatalytic activity. The composite not only showed excellent stability but also increased electrocatalytic activity, as shown by electrochemical experiments. In addition to an overpotential of 105 mV at 10 mA cm⁻² for the hydrogen evolution process, the Ni₃FeN/CC hierarchical electrocatalyst exhibited noticeably improved electrochemical characteristics for water splitting.⁸⁷ Weizhen Fang and co-workers reported an improved electrocatalyst that showed remarkable durability and efficiency for water splitting. It was composed of an array of Ni-doped cobalt di-sulfide nanowires supported on carbon cloth (Ni_{2.3%}-CoS₂/CC). The CoS₂ is well known for its great efficiency in the HER and has a crystal structure that catalyzes reactions by using Ni ions. The CoS₂ electrocatalyst enhanced by Ni not only increased HER but also sped up the OER process. This study showed that Ni-promoted CoS₂ may act as a bi-functional catalyst in the alkaline medium for both OER and HER. The Ni_{2.3%}-CoS₂/CC electrode, in the end, reached a current density of 100 mA cm⁻² at a 231 mV overpotential for HER.⁸⁸ Jieting Ding and co-workers developed a mesoporous carbon composite doped with Ni, S, and N that served as a bi-functional electrocatalyst for the OER and HER in water electrolysis. The porous structure of this composite was formed by synthesizing an N-doped porous carbon framework using NaCl as a template and integrating nickel-sulfide/nickel nanoparticles inside; the matrix was 3d and its pore size distribution was calculated using DFT. The improved HER and OER performance of Ni/NiS/NC over Ni/NC and NiS/NC was due to changed surface characteristics and chemical states, as shown by XPS analysis. Effective mass diffusion and the elimination

of hydrogen and oxygen gasses produced during electrolysis were made possible by the mesoporous structure and large surface area of the carbon matrix. Enhancing electrocatalytic efficiency and preventing particle aggregation were two benefits of adding transition metal compounds into the carbon matrix. The most active catalyst produced in this study was very efficient for water electrolysis, as evidenced by its low HER overpotential of 70 mV at 10 mA cm⁻² and Tafel slope of 45 mV dec⁻¹, shown in Fig. 8b and c.⁸⁹ Muhammad Imran Abdullah and co-workers developed a simple method for producing Ni-doped Co hollow nanoparticles (CoNi-HNP) by combining non-3d high-valence metal tungsten with non-metallic additives (B and P) in an ultrasound-assisted chemical reduction process. Using this method, the HER adsorption energy was maximized. As the most effective non-noble metal electrocatalyst for hydrogen production in neutral solutions, CoNi-HNP outperformed Pt/C in terms of stability and activity during prolonged electrolysis durations. In neutral pH electrolytes, CoNi-HNP notably reached a current density of 10 mA cm⁻² at a low overpotential of 41 mV. The amounts of P and B additions had a major impact on the catalytic activity of CoNi-HNP. Low overpotential allowed for high current densities and consistent performance with heavy mass loading over extended electrolysis periods.⁹⁰ Yaqian Yang and co-workers reported Ni-doped MoSe₂ nanosheets with Ni-Se bonds to improve alkaline electrocatalytic hydrogen evolution. Using an ethanol solution, NiO₂ was evenly distributed over MoSe₂. After annealing, the NiO₂ nanoclusters became approximately 4 nm in size and were attached to the MoSe₂ nanosheets with the help of Ni-Se bonds. DFT calculations revealed that additional catalytic sites were formed at dopant atoms or adjacent Mo- or Se atoms. XPS revealed the formation of Ni-Se bonding and lower valence. Ni species indicated electronic modulation, which led to proton reduction and increased HER activity. MoSe₂ has potential as a catalyst for hydrogen evolution, but its large overpotential due to its low active site density limits its practicality. While Ni was deposited at a rate of 4.5 at% in 1 M KOH, MoSe₂ nanosheets' active sites were increased and the overpotential was reduced to 181 mV at 10 mA cm⁻². Due to the Ni-Se bond's development, Ni-doped MoSe₂ also showed outstanding stability over 12 h.⁹¹ In both acidic and alkaline conditions, Ni-doped CoP/Co₂P nanospheres have been reported by Tianyun Chen and co-workers to be highly effective and stable catalysts for hydrogen evolution. Ni-induced modifications to electron valence states, which worked in concert with Co atoms to improve catalyst efficiency, were responsible for the enhanced catalytic activity and long-term stability of Ni-doped CoP/Co₂P nanospheres. Ni-doped CoP/Co₂P nanospheres preserved their shape during electrolysis, which helped speed up mass and electron transport and increased overall stability. A multitude of active catalytic sites were exposed, and rapid hydrogen evolution was promoted by the synergistic interaction between Ni and CoP/Co₂P, which optimized the hydrogen adsorption energy. A current density of 10 mA cm⁻² was attained by this catalyst at an overpotential of 100 mV.⁹² In both acidic and alkaline conditions, Ni-doped



FeP/C hollow nanorods have been reported by Yan Zhou Lu and co-workers to be highly effective and stable catalysts for hydrogen evolution, shown in Fig. 8d. Ni-induced modifications to electron valence states, which work in concert with Co atoms to improve catalyst efficiency, were responsible for the enhanced catalytic activity and long-term stability of Ni-doped FeP/C hollow nanorods. These Ni-doped FeP/C hollow nanorods preserved their shape during electrolysis, which helped speed up mass and electron transport and increased overall stability. DFT calculations provided insight into the catalytic role played by nickel doping at the atomic level. XPS study showed that the catalyst's surface was oxidized to phosphates, that electrons were transferred from P to Fe as demonstrated by a negative shift for Fe 2p and a positive shift for P 2p shifts, and that Ni doping increased phosphide content. A multitude of active catalytic sites were exposed, and a fast evolution of hydrogen gas was encouraged by the synergistic impact of Ni and FeP/C, which lowered the energy needed for hydrogen absorption. A current density of 10 mA cm⁻² was attained by this catalyst at an overpotential of 95 mV in alkaline medium.⁹³ The Ni-doped HER electrocatalysts are summarised in Table 4.

4.1.4 Mo-doped electrocatalysts for HER. Molybdenum doping influences morphology, conductivity, and interaction with other transition metals to maximize HER activity and stability, and molybdenum doping in the HER has shown promise as a means of improving catalytic efficiency. As with other non-noble metal doping metals, the quantity of Mo doping influences catalytic performance as well. Therefore, the precise amount of doping needs to be carefully considered when designing catalysts. Mo doping also has a major effect on the catalyst's shape, promoting the formation of structures with large specific surface areas. On the other hand, too much Mo content may cause phase dissociation and reduce the activity of the catalytic process. As a result, it is difficult to precisely control Mo doping levels for a uniform distribution and ideal concentration. Moderate metal-H bonding might promote positive HER under such conditions. Furthermore, Mo improves the catalytic activity of carbides and provides pH-universal electrolysis with remarkable stability and characteristics. The observed improvement in HER activity is probably due to the synergistic action of Mo dopants together with

other 3D transition metals. Mo dopants also result in exceptional electrocatalytic activity with low overpotential due to their porous design, enhanced electric conductivity, and cooperative effects with other transition metals.^{96,97,106}

3D-arranged molybdenum (Mo)-doped nickel sulphide (Ni₃S₂) particles have been synthesized by Chengrong Wu and co-workers as an efficient dual-electrocatalyst for the overall splitting of water. This work aimed to successfully synthesize Mo-doped nickel sulfide (Ni₃S₂) three-dimensional (3D) curved nanosheets on Ni-foam using a one-step hydrothermal technique. Furthermore, the three-dimensional (3D) hierarchical structure provided a larger contact surface area for electrolyte adsorption and desorption. Mo-inclusion had a significant impact on the structure and shape of Ni₃S₂ nanosheets, shown in Fig. 9a. Additionally, when the electrode's surface and roughness grew due to Mo element doping, more electrons from the electrolyte solution were adsorbed. The catalyst had exceptional electrochemical performance concerning HER. At 10 mA cm⁻², it displayed a low overpotential of 212 mV, which is a significant drop of 127 mV compared to un-doped nickel sulphide.⁹⁶ By the use of a two-step approach, Yiqiang Sun and co-workers have successfully synthesized Mo-incorporated Ni₂P nanowires (NWs) on a Ni foam (NF) substrate, demonstrating their utility as a steady and strong electrocatalyst for HER throughout the pH spectrum (0–14). Molybdenum addition considerably increased the catalytic activity of Ni₂P, according to electrochemical analyses. After 24 h of HER operation at different pH settings, Mo–Ni₂P NWs/NF demonstrated outstanding durability with low activity destruction. Overpotentials of 67 mV in acidic, 78 mV in alkaline, and 84 mV in neutral environments were needed, as well, to reach a current density of 10 mA cm⁻². The remarkable HER performance can probably be ascribed to the mutually beneficial interaction between the atoms of Ni (nickel) and Mo (molybdenum).⁹⁷ Zhengnan Wang and co-workers used a single-step electrodeposition approach to produce a Mo-incorporated Ni–Cu catalyst for HER catalysis. The catalyst's optimum composition of NiCuMo, with (8% of Mo-content in solution) showed better HER performance than Ni–Mo and Ni–Cu catalytic compounds that were binary, owing to the three metals' synergistic connection. Using the strong molybdenum–H and weak nickel–H binding, the combination of nickel and molybdenum

Table 4 Ni-doped electrocatalysts for HER

S. no	Catalyst	Electrolyte	Overpotential (η) at 10 mA cm ⁻²	Tafel slope (mV dec ⁻¹)	C_{dl} (mF cm ⁻²)	Stability (hours & cycles)	Ref.
1	Ni–Co–MoS ₂	0.5 M H ₂ SO ₄	155	51	11.7	1000 CV cycles	84
2	Ni–FeP/TiN/CC	1 M KOH	71	73	—	10 h	85
3	NiMo ₂ C/NF	—	150 at (100 mA cm ⁻²)	36.8	—	10 000 CV cycles	86
4	Ni ₃ FeN/CC	1 M KOH	105	61	52.55	20 h	87
5	Ni _{2.3%} –CoS ₂ /CC	1 M KOH	231 at (100 mA cm ⁻²)	106	—	12 h	88
6	Ni/NiS/NC	1 M KOH	70	45	22.19	3000 CV cycles	89
7	CoNi–HNP	1.0 M PBS	41	35	130	20 h	90
8	Ni–MoSe ₂	1 M KOH	181	83	21	12 h	91
9	Ni–CoP/Co ₂ P	1 M KOH	125	73	4.4	12 h	92
10	Ni–FeP/C	1 M KOH	70	54	80.2	12 h	93



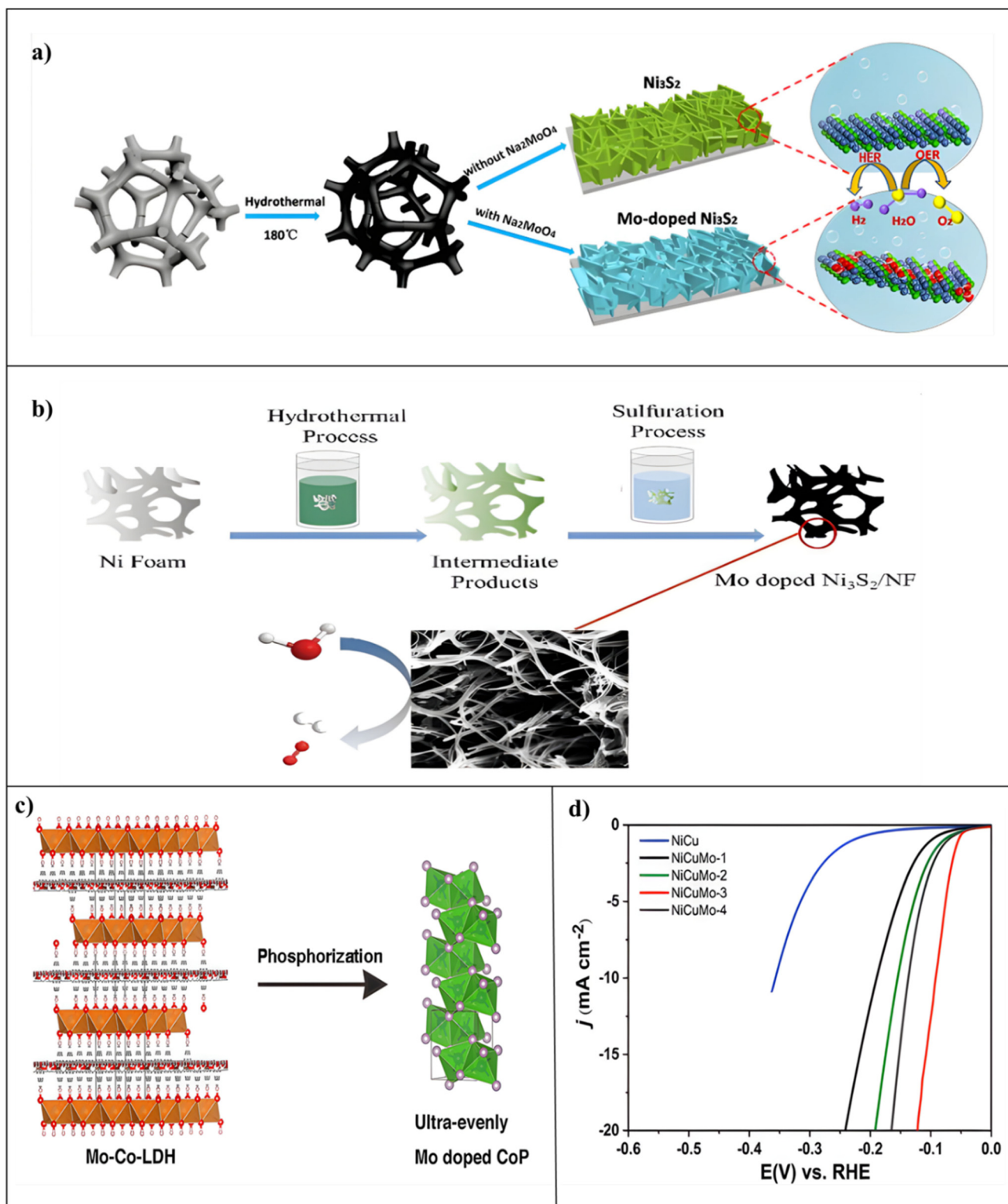


Fig. 9 (a) A demonstration of the Mo-doped Ni_3S_2 nanosheet design. Reproduced from ref. 96 with permission. Copyright 2018, Elsevier. (b) On nickel foam (NF) a 3D Mo-doped Ni_3S_2 nanowire network was successfully produced using a hydrothermal and sulfidization process. Reproduced from ref. 103 with permission. Copyright 2018, Wiley. (c) Diagrammatic representation of ultra-even Mo-doped CoP composite formation. Reproduced from ref. 101 Copyright 2021, ACS. (d) NiCu and NiCuMo catalysts electrocatalytic HER performance in $0.5\text{ M H}_2\text{SO}_4$. Reproduced from ref. 98 Copyright 2019, RSC.



enhanced HER activity in alkaline environments. Additionally, in comparison with the Ni–Mo catalyst the addition of copper greatly assisted in the production of a thick solid nanoparticle shape. In an alkaline electrolyte, this Mo-incorporated Ni–Cu electrode (8% Mo-content in solution) outperformed both bimetallic Ni–Mo and Ni–Cu catalysts, showing excellent electrochemical resistance and a minimal overpotential of 95 mV at a current density of 10 mA cm⁻². Bimetallic Ni–Mo and Ni–Cu catalysts showed excellent electrochemical stability and a minimal overpotential of 95 mV at a current density of 10 mA cm⁻², shown in Fig. 9d.⁹⁸ Yan Zhou and co-workers reported that molybdenum loading induced metallic characteristics in Co–Se by enhancing the electrocatalytic synthesis of hydrogen. The existence of a mixed phase of Co₉–Se₈ and Co–Se was required for the effective integration of molybdenum. Molybdenum doping successfully tuned the crystalline and electronic configuration of Co–Se, causing lattice-contracting and metallic properties that enhanced its electrocatalytic performance for the HER. This was confirmed by computational models, which showed that molybdenum doping caused lattice contraction in Co–Se and caused the electron density of states to change in the direction of the Fermi level. The reduced cell volume and lower Co oxidation state were compatible with XPS and XANES studies, demonstrating that a small Mo substitution may compress the lattice despite Mo having a higher atomic radius than Co. Despite a small slope of Tafel of 58.7 mV dec⁻¹ and the greatest HER activity of every substance examined, Co_{0.8}–Mo_{0.2}–Se required only an 186.1 mV overpotential to reach a current density of 100 mA cm⁻².⁹⁹ An efficient and long-lasting electro-catalyst for water splitting was synthesized by Zheng Cui and co-workers in their work, an optimized synthesis of Mo-loaded Ni₃–S₂ nano-rods. As an instance of the way the atomic stoichiometric proportions of molybdenum and nickel may be controlled by varying the temperature of the reactions, they showed that Mo-doped Ni₃–S₂ can grow directly on nickel foams using sodium molybdate (Na₂MoO₄) as the molybdenum source at different temperatures. DFT simulations and experimental investigations both demonstrated that the under-coordinated Mo–S sites on the outermost layer of MoS₂ have a high capacity for the hydrogen chemisorption process. XPS measurements were used to identify and characterize prominent constituent elements and their chemical valence states. A mild 278 mV overpotential at a 100 mA cm⁻² hydrogen synthesis current density and a small 72.9 mV dec⁻¹ Tafel slope was observed, leading to increased reactions of hydrogen evolution performance.¹⁰⁰ Liang Li and co-workers developed an *in situ* phosphorization technique using CoMo-layered double hydroxide (CoMo-LDH) as the precursor for producing extremely uniform Mo-doped CoP materials (Mo–CoP). The well-organized cation configuration in the CoMo-Layer Double Hydroxide (LDH) promoted phosphorization, which produced molybdenum that was evenly distributed throughout the Co–P framework, shown in Fig. 9c. This produced a remarkable bi-functional catalyst for the total splitting of water. Even molybdenum distribution inside the Co–P framework was facilitated by the ordered cation arrangement

in the Layer Double Hydroxide nanosheet precursor, which occurred after *in situ* phosphorization. As a result, in acidic conditions, the molybdenum-doped cobalt phosphide showed increased HER activity and stability. The overpotential was just 116 mV when the current density in 0.5 M H₂SO₄ was 10 mA cm⁻².¹⁰¹ Qin Wang and co-workers report their results on creating Mo-doped Ni₂P hollow microspheres using a direct phosphorization approach in conjunction with a solvothermal process. The connected Gibbs free energy of H* absorbance (ΔG_{H^*}) dropped from 0.33 eV (P site) to 0.21 eV (bridge Mo–Ni site) after molybdenum doping, with the active centre transitioning to the molybdenum–nickel bridge site. The reduced ΔG_{H^*} of this ideal active site sped up the charge exchange procedure for H* intermediates and resulted in the synthesis of H₂. With a modest Tafel slope of 53.4 mV dec⁻¹ and an inadequate overpotential of 81 mV (10 mA cm⁻²), the DFT results confirmed that the bridge Mo–Ni site was the active site in the alkaline HER process, with a ΔG_{H^*} of 0.21 eV. XPS confirmed that Mo-doped Ni₂P HNs had a greater intrinsic electrocatalytic property for electrochemical oxygen generation. The Mo-added Ni₂–P hollow microspheres demonstrated exceptional HER activity in the 1.0 M KOH approach.¹⁰² Cheng Du and co-workers synthesized a technique that used a series of hydrothermal and sulfidation steps to produce 3D molybdenum-incorporated Ni₃–S₂ nanowire connections that were directly grown on nickel foam. The molybdenum-incorporated Ni₃–S₂/NF composite exhibited better catalytic activity due to its unique 3D hierarchical nanowire structures and effective molybdenum doping, shown in Fig. 9b. The inherent features of Ni₃–S₂ were synergistically enhanced by the presence of molybdenum ions (Mo⁶⁺) in the solution containing the precursor, which improved its catalytic activity toward the HER. In addition to changing the catalysts' morphology, molybdenum incorporation improved Ni₃–S₂'s electrical organization, which affected the H₂/O₂ intermediates' binding energy during the water-splitting operation. To confirm the chemical states of Mo_{5.9}Ni_{94.1}S/NF, an XPS measurement was done. The resultant 3D network, Mo_{5.9}Ni_{94.1}S/NF, exhibited impressive catalytic activity needing just 60.8 mV overpotential to achieve 10 mA cm⁻² of current density for HER.¹⁰³ The development of ultrathin and hollow Molybdenum–Nickel nanosheet arrays as extremely effective electrocatalysts for the HER has been led by Xiaoying Lu and co-workers. Since molybdenum (Mo) may induce new active sites to boost electrochemical reactions and change the crystalline makeup of composites, its distinctive physical and chemical features make Molybdenum-based materials especially appealing. Significantly, it has been shown that the bimetallic nickel–molybdenum catalysts, because of their advantageous hydrogen binding power and greater corrosion resistance, systematically show higher activity than pure nickel-based materials. The resultant Mo–Ni nanosheets had a modest Tafel slope of 62 mV dec⁻¹ and an overpotential of 91 mV at 10 mA cm⁻², indicating improved HER effectiveness.¹⁰⁴ Molybdenum doping-induced improvements in the performance of urchin-like W₁₈O₄₉ nanostructures for the hydrogen evolution process have been developed by Xing



Zhong and co-workers. Because the molybdenum dopant modifies associated chemical characteristics, more active sites develop, which accounts for Mo-W₁₈O₄₉'s improved water-splitting performance. Furthermore, the exposed large surface area of the urchin-like form decreased electronic transmission limits. These results were corroborated by density functional theory computations, which showed that the Mo dopant's capacity to increase the number of active sites, optimizing the adsorption of hydrogen owing to electrical and geometric modification, is what allowed for the notable performance improvement. Additionally, the 3-dimensional urchin-like shape with a lot of 1-dimensional nanowires facilitated electron transmission and ensured quick interfacial exchange of charges to improve electrocatalytic processes. The inherent enhancement in HER activity with Mo-W₁₈O₄₉ was demonstrated by both empirical and theoretical findings. With a modest starting potential of 23 mV and a Tafel slope of 54 mV dec⁻¹, the resulting 3D urchin-like Mo-W₁₈O₄₉ showed outstanding electrocatalytic activity for HER.¹⁰⁵ The Mo-doped HER electrocatalysts are summarised in Table 5.

4.1.5 Mn-doped electrocatalysts for HER. From the optimum use of its unique electrical and structural properties, manganese doping in the HER offers the potential for enhancing its catalytic effectiveness and stabilization, hence promoting the emergence of sustainable and effective techniques for producing hydrogen. The catalytic potential of transition metal-based catalysts is effectively activated by manganese doping, which results in lower $|\Delta G_{H^*}|$ values. More active sites are generated by the addition of Mn doping, which causes mild atomic deformations and activates the HER.^{107,108} During catalysis, Mn-doped non-noble metal phosphides also show much lower $|\Delta G_{H^*}|$ levels and overpotential over the pH range. Nevertheless, difficulties like phase dispersion and the ambiguity of the linkages between catalyst structures, doping concentrations, and electrocatalytic capabilities still exist.¹⁰⁹

Manganese-doped Ni-P₂ particles on carbon cloth (Mn-Ni-P₂ NSs/CC), which function better than pure Ni-P₂ in three dimensions and are more stable at all stages of pH, have been successfully synthesized by Xiaodeng Wang and co-workers, shown in Fig. 10a. The addition of Mn increased the catalytic activity of NiP₂ from electrochemical analyses. With a high phosphorus material, the Mn-Ni-P₂ NSs/CC acted as a poly-

phosphide and provided many active sites that were favorable to catalytic activity. Increased specific surface exposure to catalytically active sites was provided by the 3D nano-sheet structure of Mn-Ni-P₂ NSs/CC, which further facilitated electron transport. According to calculations using DFT, the presence of Mn elevated the electrocatalytic performance by lowering the kinetic energy limit for H atom adsorption on the surface of Ni-P₂ NSs. XPS studies indicated that Mn doping led to a reduced transfer of electron density from Ni to P, potentially improving HER performance. It took overpotentials of 97 mV in 1.0 M KOH to get the same current density.¹¹⁰ Mn incorporation into pyrite CoSe₂, resulting in minor atomic rearrangements to provide more active edge sites for the HER, has been reported by Youwen Liu and co-workers. Using DFT simulations, it was discovered that Mn inside the CoSe₂ lattice lowers the kinetic energy barrier and promotes the formation of H-H bonds between neighbouring adsorbed hydrogen atoms, thereby improving H₂ generation. CoSe₂ ultrathin nanosheets doped with Mn provided a way to modify the electronic structure for HER dynamics that are optimal. Considering an exchange current density of 68.3 μA cm⁻², a quite modest Tafel slope of 36 mV dec⁻¹, and a very low overpotential of 174 mV, these Mn-doped CoSe₂ ultrathin nanosheets indicated an impressive feat in HER performance.¹¹¹ Tingting Liu and co-workers reported the first development of Mn-loaded Co-P (Mn-Co-P) nanosheets arranged on titanium mesh (Mn-Co-P/Ti) as an effective three-dimensional electrode for the hydrogen evolution process with strong stability at all different pH values. Density functional theory simulations indicated that Mn-Co-P had a higher thermo-neutral hydrogen energy than Co-P, which accounted for its increased HER activity. As the ΔG_{H^*} gets closer to thermoneutrality, the catalytic activity of HER is enhanced. Further investigation of the electrical structures showed that neighbouring cobalt and phosphorus atoms get electron donations from doped Manganese atoms. The XPS spectrum of Mn-Co-P indicated the presence of Mn, Co, and P elements. Signals of C and O elements may be due to product contamination or surface oxidation. According to a Bader charge analysis, nearby cobalt atoms get up to 0.38 electrons from nearby Manganese atoms. The Mn-Co-P nano-array integrated on titanium mesh required overpotentials of 49 mV and 86 mV in 0.5

Table 5 Mo-doped electrocatalysts for HER

S. no	Catalyst	Electrolyte	Overpotential (η) at 10 mA cm ⁻²	Tafel slope (mV dec ⁻¹)	C_{dl} (mF cm ⁻²)	Stability (hours & cycles)	Ref.
1	Mo-doped Ni ₃ S ₂	1 M KOH	212	98	4.02	1000 CV cycles	96
2	Mo-Ni ₂ P NWs/NF	1 M KOH	78	109	—	24 h	97
3	NiMoCu-3	1 M KOH	90	41	5.1	5000 CV cycles	98
4	Co _{0.8} Mo _{0.2} Se	0.5 M H ₂ SO ₄	186 at (100 mA cm ⁻²)	58.7	—	10 h	99
5	200-SMN/NF	1 M KOH	278 at (100 mA cm ⁻²)	72.9	—	—	100
6	CoMo-LDH	1 M KOH	118	69	—	24 h	101
7	Mo-doped Ni ₂ P HNs	1 M KOH	81	53.4	38	10 h	102
8	Mo _{5.9} Ni _{94.1} S/NF	1 M KOH	61	39.2	11.4	—	103
9	Mo-Ni-NSs-3	1 M KOH	91	62	30	50 h	104
10	10% Mo-W ₁₈ O ₄₉	0.5 M H ₂ SO ₄	45	54	5.9	5000 CV cycles	105



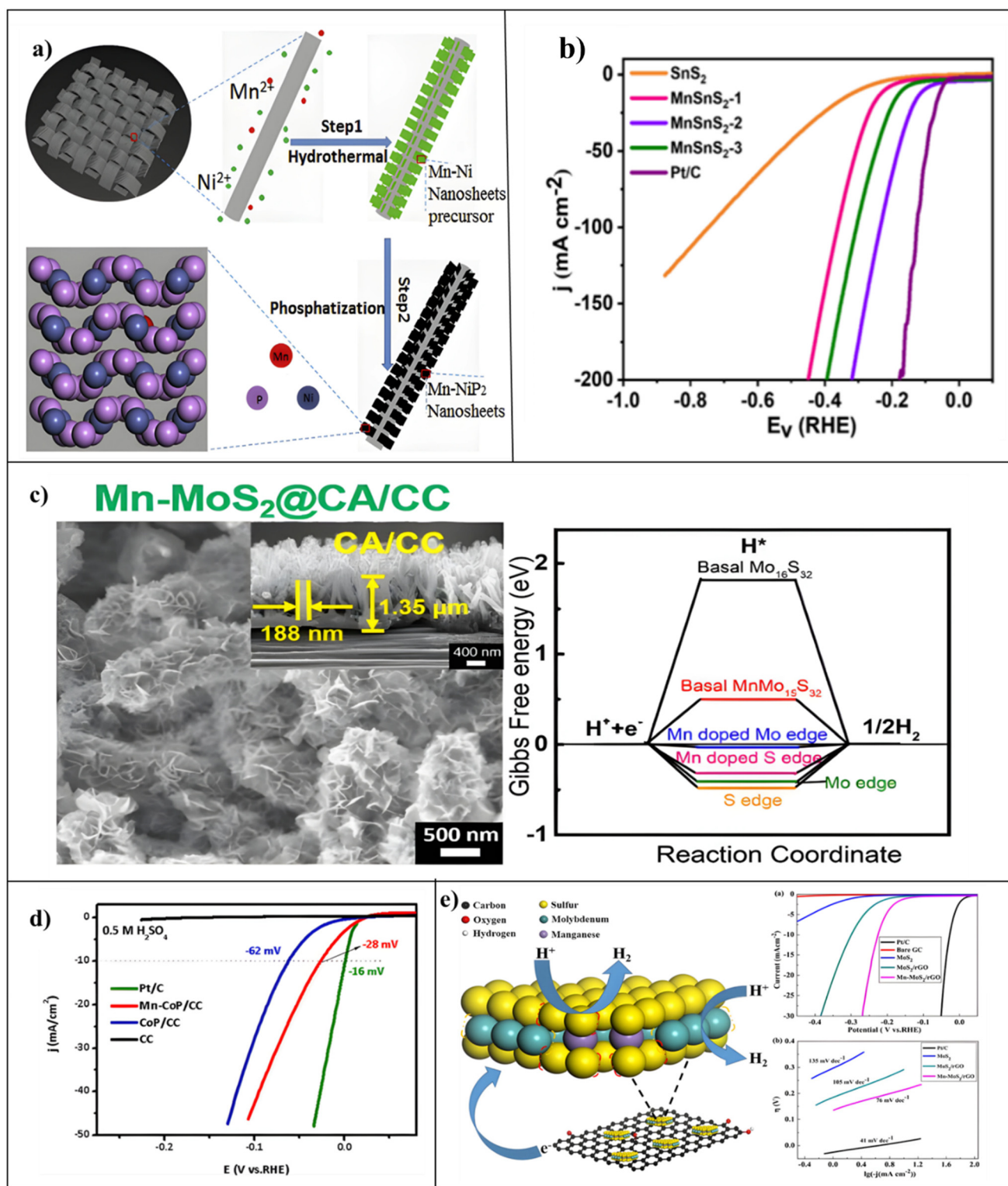


Fig. 10 (a) Diagram showing the steps involved in creating manganese Mn-doped NiP₂ nanosheets over carbon cloth. Reproduced from ref. 110 with permission. Copyright 2018, Elsevier. (b) A demonstration of the process used to prepare the MnSnS₂-2. Reproduced from ref. 112 with permission. Copyright 2018, RSC. (c) TEM and HRTEM pictures of the 0.5 Mn-MoS₂ compound as well as a hydrogen ΔG_{H^+} diagram showing the doped and doped-free MoS₂. Reproduced from ref. 115 with permission. Copyright 2018, ACS. (d) The LSV curves for HER in 0.5 M H₂SO₄ for Mn-CoP/CC, CoP/CC, pure CC, and Pt/C catalysts at a scan rate of 1 mV s⁻¹. Reproduced from ref. 116 with permission. Copyright 2020, Elsevier. (e) The Mn-MoS₂/rGO ball model. Reproduced from ref. 117 with permission. Copyright 2017, Elsevier.



M H₂SO₄ and 1.0 M KOH, respectively, to get a current density of 10 mA cm⁻².¹⁰⁸ Athibala Mariappan and co-workers studied Mn@SnS₂-2 to improve the hydrogen generation process. They described a simple doping approach for developing a S vacancy-rich Mn@SnS₂ electrocatalyst in alkaline medium (1 M KOH). The as-synthesized Mn@SnS₂-2 catalyst exhibited outstanding bifunctional electrochemical performance, with overpotentials of 260 mV (OER) and 108 mV (HER) at 10 mA cm⁻².

The preparation process for the MnSnS₂ heterostructure material is given in Fig. 10b. The improved catalytic efficiency of the Mn@SnS₂-2 catalyst is due to the development of numerous Mn dopants generating sulfur defects. Doping Mn into the SnS₂ crystal structure increases active sites, facilitates charge transfer, and improves intermediate adsorption and desorption.¹¹² Min Wang and co-workers developed a simple and regulated synthesis method for a mesoporous Manganese-doped Fe-P (Mn-FeP) catalyst for the hydrogen evolution process that is both highly effective and persistent. Mesoporous Fe-P undergoes a broad pH range electroactivity boost due to Manganese doping. Topology engineering and heteroatom implantation techniques work together to maximize the reaction kinetics for HER by increasing the specific surface area of Fe-P and altering its electronic makeup. Based on physical/electrochemical analysis and DFT calculations, it was determined that adjacent doped Mn atoms with weaker electron negativity provided partial electrons to the Fe atoms in Mn-FeP. This resulted in free energy for thermoneutral hydrogen adsorption on active sites, improving intrinsic activity. By utilizing a template-free technique to increase the specific surface area, this approach avoided the problems associated with traditional template approaches such as synthesis that is unpredictable and tedious, and ensured the effective doping of Manganese into the phosphide. With overpotentials of 69 mV and 173 mV in H₂SO₄ and KOH electrolytes, respectively, at 10 mA cm⁻², the mesoporous Mn-FeP catalyst demonstrated outstanding activity and stability for HER over an extended pH range.¹⁰⁹ The synthesis of Mn-doped Ni₂-P nanosheet arrangements on nickel foam (NF) (Mn-Ni₂-P/NF) was reported by Ya Zhang and co-workers, and this catalyst served as extremely effective electrodes for the hydrogen evolution process in alkaline environments. The electrical driving capacity of Mn-Ni₂-P/NF in 1.0 M KOH was 20 mA cm⁻² at an overpotential of 103 V, which was 82 mV less than that required to operate Ni₂-P/NF. It also showed outstanding long-term electrochemical endurance for a minimum of 25 h. When comparing Mn-Ni₂-P/NF with Ni₂-P/NF, electrochemical impedance spectroscopy (EIS) showed a lower radius of semicircle for Mn-Ni₂-P/NF, suggesting higher electron mobility rates and quicker catalytic responses. A potential catalyst material for water-splitting devices aiming at producing hydrogen fuels on an extensive basis is provided by this study.¹¹³ Xiumin Li and co-workers have reported a technique that combines the procedures of thermal phosphorization and uni-polar pulse electro-deposition to generate Mn-incorporated cobalt phosphide (Co-P) electrocatalyst-coated electrode from Co-Mn LDH precursors as well. The inclusion of Mn signifi-

cantly improved the catalytic function and conductance of the Co-P catalysts, as shown by the results of electrochemical impedance spectroscopy (EIS) and linear scan voltammetry (LSV). The nano-sheet microstructure arranged into a porous nanoparticle cluster shape during phosphorization, converting the Co-Mn LDH precursor into Mn-CoP catalysts. Mn-CoP-coated electrodes required a significantly lower overpotential of 108 mV to keep a current density of 10 mA cm⁻² in 0.5 M H₂SO₄ solution as compared to CoP-coated electrodes.¹¹⁴ Liyang Zhang and co-workers reported the synthesis of manganese-doped Mo-S₂ ultrathin particles that were vertically oriented and fixed on a hierarchical carbon skeleton resembling fin tubes to optimize both intrinsic performance and electrode design synergistically. For enriched Mo-S₂ all the hydrogen adsorption liberated energies of the basal surfaces, sulfur edges, and molybdenum edges decreased according to computational models. Moreover, the Mn-doped Mo-S₂ monolayer's electrical composition showed a zero-band gap, a sign of enhanced intrinsic conductance. The formation of Mo-S₂ ultrathin particles surrounding carbon nanowires in a vertical orientation improved the exposure of active edge areas, which lowered the charge transport resistance and improved the durability of the 3-dimensional self-supported electrode as well, shown in Fig. 10c. Having a Tafel slope as low as 44 mV dec⁻¹, this carefully constructed, self-supported, binder-free electrode with integrated architecture showed exceptional endurance even at an elevated cathodic current density of 200 mA cm⁻² in 0.5 M H₂SO₄. It also had a modest overpotential of 130 mV at -10 mA cm⁻².¹¹⁵ An outstanding electrocatalytic activity for the HER in both alkaline and acidic conditions has been demonstrated by Siran Xu and co-workers through the development and manual production of an extremely efficient electrocatalyst with a unique tiny 3-dimensional (3D) permeable peony-like micro-flower Mn-CoP structure on adaptable carbon cloth (Mn-CoP PMFs/CC). Doping with magnesium increased electron transport and reduced the thermoneutral adsorption of hydrogen-free energy (ΔG_{H^+}), according to DFT calculations. When manganese was added to Co-P the electrical structure was altered, the number of places that were active rose, and basic nanowire arrays became permeable nanorod-like nanostructures. All these changes improved Co-P's inherent performance in HER. For 0.5 M H₂SO₄ and 1.0 M KOH approaches, the catalyst exhibited an ultra-low excessive potential of 28 mV and 90 mV as well achieving a current density of 10 mA cm⁻², shown in Fig. 10d. These results were attributed to the catalyst's unique 3D porous structure with enormous surface areas and exceptional electron conductivity for rapid mobility of electrons and fast gas release routes.¹¹⁶ The synthesis of a hybrid catalyst using Mn-doped MoS₂/reduced graphene oxide (Mn-MoS₂/rGO) was accomplished by Liqian Wu and co-workers using the hydrothermal approach. This catalyst's exceptional HER performance was mostly due to reduced graphene oxide's excellent electrical conductivity, but it also greatly benefitted from the basic catalytic function boost that comes from adding magnesium Mo-S₂. XPS measurements were performed to study the chemical compo-



sition of the Mn–MoS₂/rGO hybrid. With a minimum Tafel slope of 76 mV dec⁻¹, significant cathodic currents, and a tiny overpotential of 110 mV, the Mn–MoS₂/r-GO hybrid catalyst demonstrated remarkable HER activity, shown in Fig. 10e.¹¹⁷ The Mn-doped HER electrocatalysts are summarised in Table 6.

4.1.6 Cu-doped electrocatalysts for HER. Using copper (Cu) as a dopant or a composite material component can accelerate the HER process. Moreover, Cu may modify the electrical structure of the doped material (such as carbon-based materials or transition metal oxides). Improved catalytic activity in the direction of the HER could arise from this modification. To be more precise, Cu can improve the overall reaction by modifying the binding energies of reaction intermediates involved in the HER, such as adsorbed species and hydrogen atoms. Additionally, Cu incorporation can increase the number of active spots on the material's surface. At these sites, hydrogen atoms may more easily bond and react to generate hydrogen gas, acting as nucleation sites for hydrogen evolution. The HER process is now more efficient as an outcome. Alongside other metals or materials like Pt, Ni, or carbon-based compounds like graphene, Cu can produce heterostructures or alloys. These hybrid structures may demonstrate synergistic effects, in which the combined materials outperform the individual components in terms of catalysis. In conclusion, Cu can reduce the activation energy needed for the HER by improving the kinetics of the reaction and promoting electron transport pathways. This implies that it is possible to accelerate the rate at which hydrogen gas develops off the catalyst surface, resulting in higher total HER rates of reaction.^{118–120}

Cu-doped Ni₃S₂ interconnected nano-sheet arrays are very effective electrodes for improving alkaline hydrogen development, and were developed by Yangyang Ding and co-workers. However, because of the intense S–H bond connection on the outermost layer of Ni₃S₂, there is still room for improvement in the electrochemical reactions of the cathode-reducing process. The 2-step hydrothermal technique was used to produce an extensive range of non-precious metal Cu–Ni₃S₂/NF hybrids to solve this specific issue. Copper and nickel provide a good basis for introducing copper into Ni₃S₂ since they are neighbouring elements in the same period with comparable electro-negativity and radius of atoms. Copper doping, which gener-

ates morphological modifications, raised the number of electrochemically active sites and improved the intrinsic activity of the catalyst itself, and was directly responsible for the outstanding characteristics and improved efficiency of the HER. Experimental results and DFT calculations revealed that Cu–Ni₃S₂/NF outperforms the other samples. Under a low overpotential of 92 mV, the copper-doped material enabled current densities of 10 mA cm⁻² during the HER process.¹²¹ For the HER, Qiangqiang Sun and co-workers have developed a new 3-dimensional hierarchically arranged nano-tubular copper-added nickel catalyst on nickel foam (Ni(Cu)/NF). The catalyst was synthesized using a simple procedure that included electro-deposition and selective electrochemical dealloying processes. Density functional theory simulations showed that a more favourable energy state for the adsorption of hydrogen was produced by copper replacement and the production of nickel oxide (Ni–O) on the outer layer. A considerably improved catalytic efficiency was the result of the distorted lattice of nickel, caused by copper substitution, the enhanced interfacial activity brought about by oxidation at the surface of nanoporous nickel, and the abundance of sites of activity provided by the 3D hierarchical porous design. Furthermore, the interfacial function of Ni–O/Ni was improved by developing a nickel oxide shell around a portion of the Ni nanoparticle cores. This promoted hydrogen adsorption at nearby open nickel sites and increased HER activities. The catalytic efficiency was further enhanced by the significant synergistic effects of copper doping, which replaced the nickel sites and optimized ΔG_{H*} for the adsorption of hydrogen while preserving a balance between desorption and absorption. The resulting Ni(Cu)/NF electrode had an enormous electrochemical active surface area and showed Pt-like electrocatalytic properties for HER, with only a 27 mV overpotential needed to achieve a 10 mA cm⁻² current density and a 33.3 mV dec⁻¹ Tafel slope, shown in Fig. 11a.¹¹⁸ A straightforward technique utilizing a combination of hydrothermal and low-temperature phosphidation procedures for producing self-supported nanoporous copper-incorporated cobalt phosphide nanorod grids on carbon cloth (Cu–CoP NRAs/CC) with markedly increased catalytic activity has been reported by Lulu Wen and co-workers. The strong heteroatomic connections that caused many distorted lattices and defects, and hence many reactive areas on the nano-rods,

Table 6 Mn-doped electrocatalysts for HER

S. no	Catalyst	Electrolyte	Overpotential (η) at 10 mA cm ⁻²	Tafel slope (mV dec ⁻¹)	C _{dl} (mF cm ⁻²)	Stability (hours & cycles)	Ref.
1	Mn–NiP ₂ NSs/CC	1.0 M KOH	97	45	64.3	20 h	110
2	Mn-incorporated CoSe ₂	0.5 M H ₂ SO ₄	174	36	—	2000 CV cycles	111
3	Mn–Co–P/Ti	1.0 M KOH	76	55	50.32	10 h	108
4	MnSnS ₂ -2	1.0 M KOH	108	41	2.56	50 h	112
5	Mn–FeP	1.0 M KOH	173	59	4.90	10 h	109
6	Mn–Ni ₂ P/NF	1.0 M KOH	103 at (20 mA cm ⁻²)	135	3.98	25 h	113
7	Mn–CoP	0.5 M H ₂ SO ₄	127	53	49.5	18 h	114
8	0.5Mn–MoS ₂ @CA/CC	0.5 M H ₂ SO ₄	130	44	40.4	2000 CV cycles	115
9	Mn–CoP PMFs/CC	1.0 M KOH	90	47.3	10.8	24 h	116
10	Mn–MoS ₂ /rGO	0.5 M H ₂ SO ₄	110	76	17.37	2000 CV cycles	117



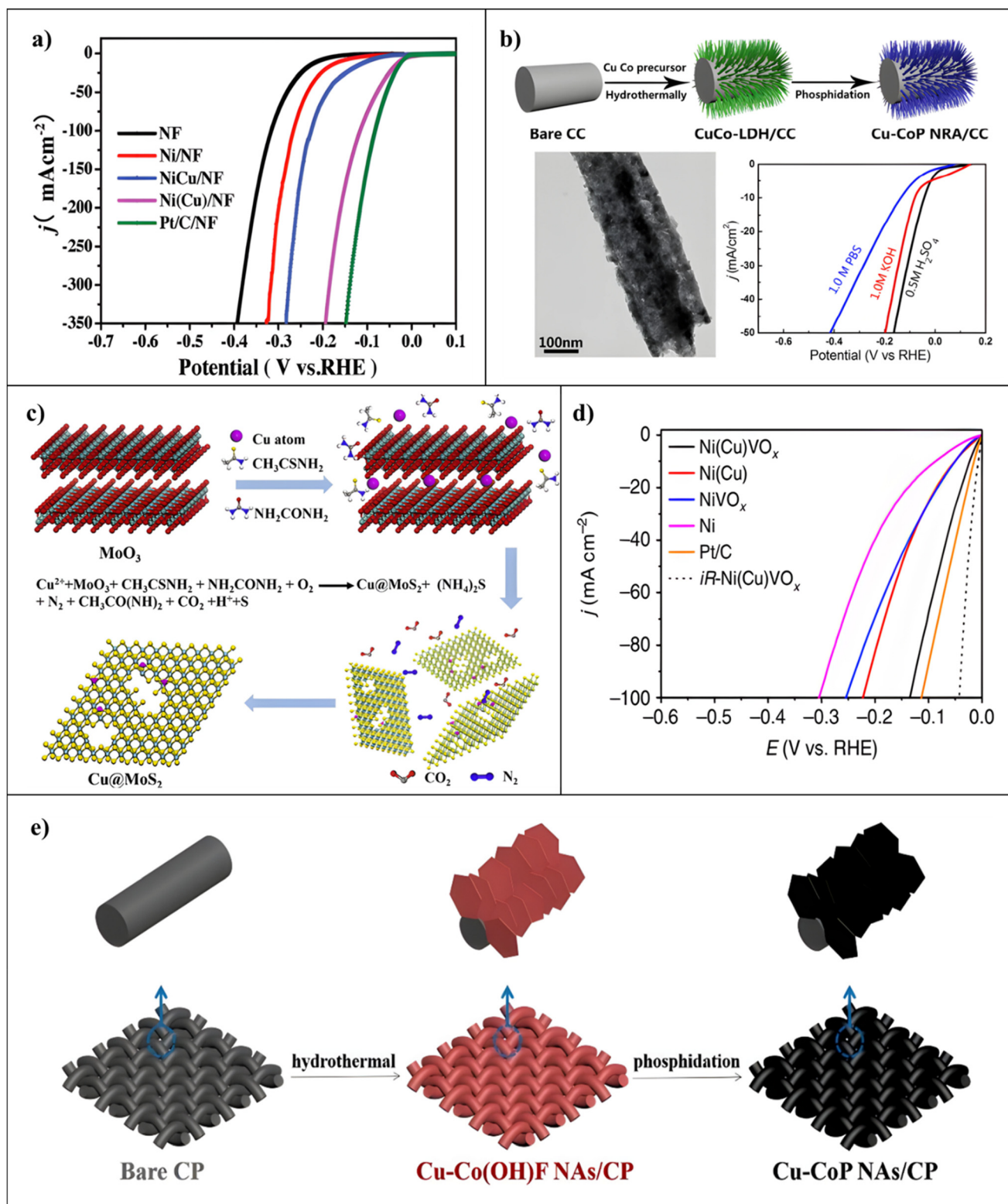


Fig. 11 (a) HER polarisation curves at 2 mV s⁻¹ with 95% iR -compensations for NF, Ni/NF, NiCu/NF, Ni(Cu)/NF, and Pt/C/NF. Reproduced from ref. 118 with permission. Copyright 2018, Wiley. (b) The Cu-CoP NRAs/CC utilized as an electrocatalyst for HER are prepared. Reproduced from ref. 122 with permission. Copyright 2018, ACS. (c) Diagrammatic representation of Cu@MoS₂ mechanism. Reproduced from ref. 123 with permission. Copyright 2019, Elsevier. (d) LSV curves for iR -Ni(Cu)VO_x. Reproduced from ref. 125 with permission. Copyright 2020, Nature. (e) Diagrammatic representation of Cu-CoP NAs/CP formation. Reproduced from ref. 129 with permission. Copyright 2019, Elsevier.



were responsible for the remarkable HER performance that Cu–CoP NRAs/CC exhibited. The kinetic energy limit for hydrogen atom adsorption on the cobalt phosphide layer was subsequently significantly lowered because of the inclusion of copper inside the cobalt phosphide framework. The study presents a new approach for synthesizing a variety of eco-friendly HER catalysts that may be used for electrochemical water splitting at any pH level. For catalytic hydrogen generation across a wide pH range, the combined Cu–CoP NRAs/CC cathode exhibited outstanding HER performance. Only 44 mV and 81 mV of overpotential were needed in 0.5 M H₂SO₄ and 1 M KOH, respectively, to reach a current density of 10 mA cm⁻², shown in Fig. 11b.¹²² Molybdenum sulfide (MoS₂) is a promising non-precious-metal catalyst option that has been reported by Liang Ji and co-workers as a cheaper alternative to platinum (Pt) catalysts in the hydrogen evolution process. Nonetheless, the restricted amount of catalytic active sites and the inert basal layer of MoS₂ significantly limit its catalytic effectiveness. This was solved by using a simple one-pot solvothermal technique for producing porous 1T-MoS₂ combined with copper atom doping. While the doping of atoms of copper (Cu) promoted electron transport between copper and MoS₂, the porous composition and 1T-MoS₂ in the resultant Cu@MoS₂ material concurrently stimulated the base layer and supplied additional active edge spots, shown in Fig. 11c. Significantly, in comparison with the 2H-MoS₂ surface, calculations based on first principles showed the larger transfer of charge as well as greater stability when copper was adsorbed on the 1T-MoS₂ surface. These findings suggest that the metastable 1T-MoS₂ monolayer may be advantageous in improving HER efficiency by adsorbing single-metal atoms. With a small overpotential of 131 mV at a current density of 10 mA cm⁻² and a modest Tafel slope of 51 mV dec⁻¹, the Cu@MoS₂ sample exhibited improved performance with HER.¹²³ Hui Su and co-workers have developed a simple method based on Ni₃S₂/Co₃S₄ to synthesize the bimetallic component sulfide heterostructured electrocatalysts (M = Cu, Fe, Mo, Zn). Requiring an overpotential of just 79 mV to attain current densities of 10 mA cm⁻² for the HER analysis, the resulting Cu–Ni₃S₂/Co₃S₄ electrode demonstrated outstanding electrocatalytic activity for HER in a 1 M KOH electrolyte. XPS study validated that the addition of copper cations promoted the development of high-valent nickel and cobalt (Ni/Co) sites. The density functional model calculations showed that by regulating the d-band centre of cobalt and lowering the adsorption energy of hydrogen and oxygen-containing intermediary states (H*, OH*, OOH*) on the surface of the catalyst, the addition of Cu boosted the formation of high-valent cobalt sites and boosted the charge transfer performance of the nickel and cobalt species, therefore improving the intrinsic catalytic capability. This work highlights the significance of transition metal ion doping for increasing the bimetallic component sulfide electrocatalytic efficiency.¹²⁴ Copper-added Nickel catalysts with Ni–O–VO_x domains integrated for the alkaline hydrogen evolution process have been reported by Yibing Li and co-workers. The development of Ni–O–VO_x sites in Ni(Cu)VO_x has been clari-

fied using systematic X-ray absorption spectroscopy investigation. The nickel framework was considerably changed by VO_x, resulting in an increased surface area that was electrochemically conductive. The super activity of the Ni–O–VO_x sites was further supported by DFT calculations based on first principles, which facilitated charge transfer from nickel to VO_x and lowered the free energy related to H-adsorption and H₂ release over nickel. This contributed to the lasting reliability and excellent HER activity of the Ni(Cu)VO_x electrode. Having an initial overpotential of almost zero and an extremely small overpotential of 21 mV, the optimized Ni(Cu)VO_x electrode achieved a current density of –10 mA cm⁻², shown in Fig. 11d, which is similar to the benchmark Pt/C catalysts.¹²⁵ A solid nickel electrocatalyst doped with copper has been demonstrated by Qiangqiang Sun and co-workers for the HER. This nanoporous catalyst possessed an ultra-large ECSA and showed outstanding activity and durability in HER. It was produced by a straightforward two-step electrodeposition–dealloying procedure over macro-porous Ni-foam. The Cu-added nanotubular nickel catalyst had several benefits that contributed to its outstanding efficiency, notably a super-large ESCA, and good hydrogen bubble dissolution as well as excellent electrical conductivity. By using the hydrogen evolution process rather than the oxygen evolution reaction, these characteristics allowed for the energy-efficient electrolytic synthesis of hydrogen. Interestingly, the Ni(Cu)/NF combination only needed a cell voltage of 0.41 V in a two-electrode alkali electrolyzer configuration to get a current density of 100 mA cm⁻² for the evolution of hydrogen.¹²⁶ Hui Li and co-workers developed a simple and efficient method for synthesizing Cu-doped bimetallic phosphosulphide nanosheet arrangements (Cu–NiS@Ni₂P/NF) on nickel foam. In the HER the CuNiS@Ni₂P/NF substance performed better, especially when the doping level of copper was optimum. Several variables contributed to the exceptional HER performance of CuNiS@Ni₂P/NF nanosheet arrays. The electrocatalyst's electronic composition and shape were effectively optimized by the moderate doping of copper. The typical configuration of nanosheet grids exposed more active places. The combined effect of various elements resulted in high intrinsic movement. Interestingly, Cu–NiS@Ni₂P/NF outperformed Cu–NiS@Ni₂P/NF-0.25 (206 mV) and Cu–NiS@Ni₂P/NF-0.125 (219 mV) with a potential of only 144 mV to obtain a current density of 10 mA cm⁻² in 1.0 M KOH.¹²⁷ Yongshang Tian and co-workers produced copper-incorporated Co₃O₄ porous nanosheet arrangements on 3-dimensional Ni-foam (Cu–Co₃O₄ NAs/NF) by straightforwardly pyrolyzing bimetal–organic structure precursors. Research findings showed that incorporating copper in Co₃O₄ materials affected the nano-sheet shape to obtain a large specific area and well-exposed places for activity. Additionally, open routes for continuous gas release and transfer of electrons were provided by the 3-dimensional electrode arrangement. Utilizing its distinct geometric and electrical structure, the Cu–Co₃O₄ NAs/NF electrode that has been developed demonstrated exceptional activity. The Cu–Co₃O₄ NAs/NF electrodes' exceptional HER activity combined with its low voltage



of 1.58 V allowed for the output of a current density of 10 mA cm⁻² for total water splitting.¹²⁸ A self-standing, non-noble metal, 3D-connected ultrathin nanosheet arrangement (Cu-CoP NAs/CP) has been introduced by Liang Yan and co-workers. The resultant Cu-CoP NAs/CP exhibited outstanding efficiency for the hydrogen evolution process because of both its improved electronic composition and higher electrical conductivity, which facilitated both the desorption and adsorption of reaction products, shown in Fig. 11e. The results of experiments and density functional theory calculations showed that the incorporation of copper to Co-P greatly improved electrical conductivity and optimized its electronic composition, which in turn facilitated the formation of O* intermediaries (which led to CoOOH active species) and H* adsorption and desorption, which occurred. This ultimately resulted in superior HER catalytic operation. The method of XPS was utilized to study the electronic structure and element states of Cu and CoP NAs. Excellent HER activity was attained using overpotentials of 81 mV to produce a current density of 10 mA cm⁻² in neutral media, owing to enhanced mass/charge transfer, optimized 3D Cu_{0.08}Co_{0.92}P NAs/CP, and greater exposure of active areas.¹²⁹ The Cu-doped HER electrocatalysts are summarised in Table 7.

4.1.7 Pt-doped electrocatalysts for HER. Platinum (Pt) is often added to HER materials or used as a dopant to greatly increase the reaction for several important factors. Platinum is well known for having outstanding HER catalytic activity. When it is placed onto or doped into other materials Pt offers active areas where hydrogen atoms may bond and undergo the essential processes to develop into hydrogen gas. Pt has ideal electrical characteristics that enable effective electron transport across the HER. This is significant because the reaction proceeds through several stages in which electrons are transferred to hydrogen ions, which form hydrogen gas in the final phase. Pt helps these electron transfer activities, which reduces the overpotential needed for the reaction to continue and raises the hydrogen evolution's general efficiency. When used for the HER in electrochemical cells, Pt is quite durable and adaptable in acidic environments. Pt can be added to composite materials or used as a dopant to increase the catalyst's stability against processes that cause degradation, such as active site aggregation or dissolution. The catalyst's long-term performance is assured by its stability, which prevents a noticeable

decline in activity. Synergistic effects can happen when Pt is mixed with other materials, including transition metals. As Pt and the host material have complementary electronic structures or optimal surface contacts, these pairings can result in improved catalytic activity as compared to Pt itself.¹⁴⁰⁻¹⁴²

Using sodium lauroyl sarcosinate (C₁₅H₂₈NNaO₃) as a structure-directing intermediary in hydrothermal conditions of use, Piyali Bhanja and co-workers have successfully synthesized mesoporous Sn-O (MTO-S). This was followed by calcination and subsequently adding Pt nanoparticles using straightforward wet-chemical techniques. The mesopore exterior of the Pt-loaded mesoporous tin oxide nanostructured materials exhibited a well-dispersed arrangement of metal and metal oxide micron-sized particles which facilitated easy electron leaping and increased the speed of the HER activity of Pt@MTO-S nanocomposites. These findings were supported by electrochemical water-splitting tests and the pore size distribution was calculated using DFT. Strong flow with H₂ bubbles over the Pt@MTO-S cathode was directly indicative of the significant potential for real-world HER implementations that Pt design on the surface of mesoporous SnO₂ possesses. Furthermore, Pt@MTO-S had an ultralow overpotential of 73 mV at 10 mA cm⁻², demonstrating remarkable HER activity.¹³⁰ The platinum-doped MoS₂ (Pt@MoS₂) catalyst that Yang Li and co-workers have reported was synthesized using a potential cycling technique. This allowed for the effective introduction of platinum dopants into the MoS₂ structure and a partial transition of MoS₂ from the 2H to the 1T stage. XPS revealed the electrical structural evolution of MoS₂ following Pt doping. The role played by platinum atoms in the electrical structural modification of MoS₂ was clarified by a thorough investigation. The theoretical calculations further demonstrated that the sulfur atoms next to Platinum in MoS₂ were the most active sites for the hydrogen evolution process, with minimal adsorption of hydrogen-free energy (ΔG_{H^*}) of around 0.04 eV, like benchmark Pt catalysts. The findings of the experiments showed that Platinum single-atom doping in 1T MoS₂ has good catalytic activity and is stable for HER. According to MoS₂'s excellent geometric and electrical framework, Pt@MoS₂ exhibited a low overpotential of 88.43 mV at 10 mA cm⁻².¹³¹ Xin Xiao and co-workers have effectively produced very low quantities of Pt-incorporated Co₂P/Ni₂P on

Table 7 Cu-doped electrocatalysts for HER

S. no	Catalyst	Electrolyte	Overpotential (η) at 10 mA cm ⁻²	Tafel slope (mV dec ⁻¹)	C _{dl} (mF cm ⁻²)	Stability (hours & cycles)	Ref.
1	Cu-Ni ₃ S ₂ /NF-1/4	1.0 M KOH	92	45.26	79.22	12 h	121
2	Ni(Cu)/NF	1.0 M KOH	27	33.3	17.27	—	118
3	Cu-CoP NRAs/CC	1.0 M KOH	81	86	113.8	40 h	122
4	Cu@MoS ₂	0.5 M H ₂ SO ₄	131	51	—	—	123
5	Cu-Ni ₃ S ₂ /Co ₃ S ₄	1.0 M KOH	79	50.4	75.4	10 h	124
6	Ni(Cu)VO _x	1.0 M KOH	21	28	—	—	125
7	Ni(Cu)/NF	1.0 M KOH	203 at (100 mA cm ⁻²)	—	—	—	126
8	CuNiS@Ni ₂ P/NF	1.0 M KOH	144	103.2	27.91	24 h	127
9	Cu-Co ₃ O ₄ NAs/NF	1.0 M KOH	125	66	11.9	20 h	128
10	Cu-CoP NAs/CP	1.0 M PBS	81	83.5	17	3000 CV cycles	129



nickel foam (Pt/Co₂P/Ni₂P/NF) by an *in situ* controlled developmental technique, demonstrating its effectiveness as an electrode for the electrolytic splitting of water into H₂, shown in Fig. 12a. The accessible Pt/Co₂P/Ni₂P/NF substrate had higher activity areas because of the super-macroporous structure of the NF substrate, which improved electron flow and provided open pathways for quick H₂ production. The enhanced pathways for actions of electrocatalytic efficiency were attributed to the complementary actions of Co₂P and Ni₂P, the number of active areas on Pt/Co₂P/Ni₂P/NF, and the control of the electronic framework by platinum doping. Considering a low overpotential of 75 mV at a current density of 10 mA cm⁻² Pt/Co₂P/Ni₂P/NF demonstrated outstanding HER activity. Furthermore, Pt/Co₂P/Ni₂P/NF ran for 25 h with remarkable stability, demonstrating exceptional durability.¹³² Sengen Anantharaj and co-workers published a technique to produce small particles of platinum colloiddally, with an average size of 3.5 ± 0.3 nm, and effectively secure them onto DNA molecules self-assemblies. A loading of 5 μL of the developed Pt@DNA homogeneous solution, or 15 μg cm⁻², was used to assess the solution straight for the electrochemical HER in 0.5 M H₂SO₄. DNA's remarkable adherence to glassy carbon and FTO substrate electrodes, together with its electrical conductivity, stability during the potentiostatic process of electrolysis, and rapid decomposition have made the produced Pt@DNA colloidal solution an excellent HER electrocatalyst. Considering the preservation of electrocatalytic activities, this technique makes it possible to manufacture ultra-small nanoparticles of platinum that have larger areas of surface without the requirement for an outside binder during the production step. In the absence of any binding agent, the Pt@DNA-GC interface displayed overpotentials of -0.026 V and -0.045 V at current densities of 10 mA cm⁻² and 20 mA cm⁻² in that order, shown in Fig. 12b.¹³³ Wenxia Chen and co-workers' design showed that adjacent platinum atomic sites contained within N-doped porous carbon nanotubes (PtSA-PtCo NCs/N-CNTs) controlled the local electronic configuration of small cobalt-platinum clusters. The combustion of zeolite imidazole MOF produced by melamine introduction with thimbleful platinum loading was used to accomplish this. Melamine was included to revive neighboring atomic carbons and start the directed development of nanotubes of carbon doped with nitrogen. A detailed investigation revealed the noteworthy influence of nearby low-coordinated Pt-N₂ sites in modifying the localized electronic configuration of tiny Pt-Co clusters. DFT calculations showed that atomically isolated PtSA could affect the electronic states of PtCo NCs, lowering the energy barrier and enhancing kinetic processes, and this led to higher electrocatalytic activity. XPS was used to analyze the chemical compositions of PtSA-PtCo NCs and PtCo NCs/N-CNT-900 samples. The performance of the HER was further enhanced by optimum control of the electronic configuration by nearby platinum atoms, which lowered the reaction energy limit in the electrochemical step. In 1 M KOH, the optimized PtSA-PtCo NCs/N-CNTs-900 demonstrated remarkable HER catalytic activity, attaining a current density of 10 mA cm⁻² and a modest over-

potential of 47 mV.¹³⁴ Huiting Hu and co-workers developed a simple hydrothermal approach for designing MoS₂ nanosheets that were deposited on Pt-incorporated yeast cell biomass (MoS₂@Pt/YC). The primary challenge was to reduce the noble metal loading without sacrificing efficiency. The nanosheets of MoS₂ were encouraged to develop on yeast cells' smooth surface, and metal Platinum could bind to its functional groups as well. As a result, nanosheets of MoS₂ were applied to the Pt/YC material, improving the exposure of the hydrogen synthesis reaction's active sites (HER). Precious metal importing can be reduced without compromising qualities by adding transition metal sulfides with a considerable amount of specific surface area. This work provides a straightforward approach for the production of Pt-doped HER electrodes. Despite a tiny Tafel slope of 74 mV dec⁻¹ demonstrating its outstanding HER efficiency and an overpotential of 118 mV at the benchmark current density of 10 mA cm⁻², the resultant MoS₂@Pt/YC material exhibited outstanding results, shown in Fig. 12c.¹³⁵ Yan Sun and coworkers have presented a simple, adaptable N₂-plasma approach for facilitating the transformation of MoS₂ nanosheets from the 2H stage to the 1T stage with a remarkable conversion rate of almost 62%. XPS was employed to study the phase transition and composition of N-MoS₂ and N,Pt-MoS₂ nanosheets from 2H to 1T. In addition, the plasma processing promoted the decomposition of Platinum salts that had been deposited and the dispersion of platinum atoms into MoS₂ nanosheets, which increased the 1T stage yield to around 87%. According to computational models and structural analysis, MoS₂'s electrical and coordination properties were successfully modulated by the deep doping of nitrogen and platinum atoms, shown in Fig. 12d. This activation of the sulfur spots resulted in the generation of emptier 2p_z orbitals that were favourable for the adsorption of water and hydrolysis. Interestingly, the method for plasma production described in this study could be easily expanded to provide large-area N, Pt MoS₂ nanosheet-coated carbon cloth that was 32 cm by 16 cm in size, with a homogeneous 1T-phase feature and a high capacity to generate hydrogen as well. The resultant Pt-MoS₂ nanosheet complexes mounted on carbon cloth showed good alkaline HER activities, obtaining strong cycle endurance and a modest overpotential of 38 mV at 10 mA cm⁻².¹³⁶ A. Abdolmaleki and co-workers developed a tiny material made of sulfonated material graphene and poly(3,4-ethylene dioxythiophene) (PEDOT) conductive polymer adorned with platinum-based nanoparticles (PtNPs). PEDOT promoted proton diffusion on the surface of the electrode, which helped to facilitate the hydrogen synthesis process (HER) in addition to avoiding nanoparticle aggregation. Increased modification of graphene nanosheets with the addition of sulfonate moieties could further improve the efficiency of the electrodes. After 500 cycles, the resultant HER electrocatalyst showed negligible overpotential (around -0.040 V vs. RHE), modest charge transfer resistance (about 50 ohms), and steady efficiency. Tafel kinetics regulated the electrocatalytic process mainly; electrochemical adsorption-desorption was important because of kinetic factors (such as Tafel



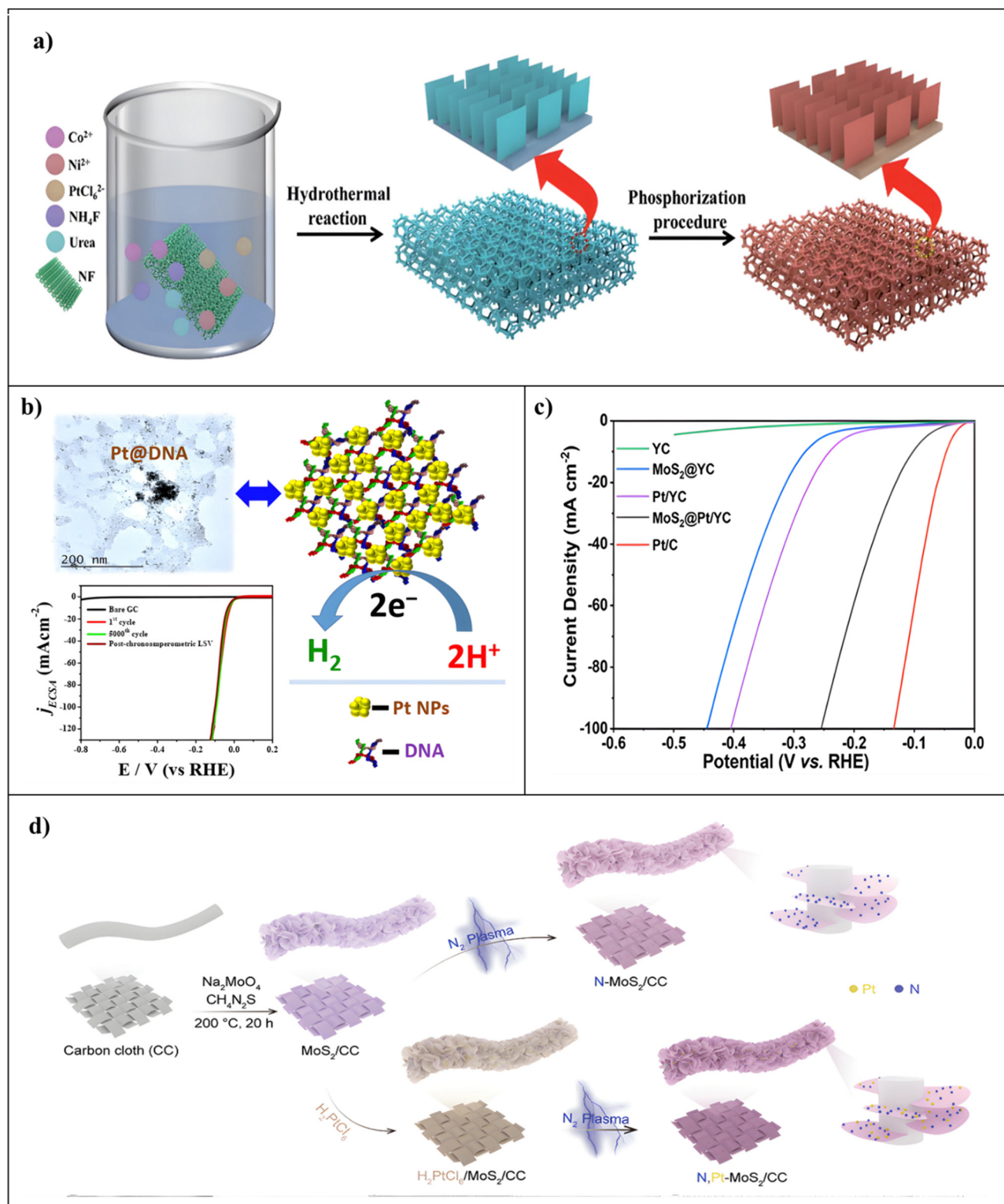
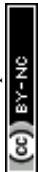


Fig. 12 (a) An illustration of the Pt/Co₂P/Ni₂P/NF synthetic process. Reproduced from ref. 132 with permission. Copyright 2020, RSC. (b) A representation of colloidal Pt@DNA molecular self-assembly general production for HER application. Reproduced from ref. 133 with permission. Copyright 2016, ACS. (c) LSV curves in H₂SO₄ (0.5 M) for Pt/C, YC, MoS₂@YC, Pt/YC, and MoS₂@Pt/YC. Reproduced from ref. 135 with permission. Copyright 2023, ACS. (d) Synthesis of nanosheets of N, Pt–MoS₂. Reproduced from ref. 136 with permission. Copyright 2022, RSC.



slope of 28.4 mV dec⁻¹).¹³⁷ Aixin Fan and coworkers have reported a simple method for synthesizing ultralow (0.66 wt%) Pt-doped FeNi metallic NPs that were embedded in a hollow graphene shell construction, or Pt-FeNi@C. Many core-shell particles were arranged during the construction of Pt-FeNi@C to prevent the Pt-FeNi metallic core from aggregating and dissolving during electrocatalytic processes. The hollow carbon shell also sped up electron transport, which greatly improved overall electrochemical efficacy. At an overpotential of 50 mV, Pt-FeNi@C notably showed a mass activity 63.0 times greater compared to that of the commercial Pt/C catalyst. The unique structural and electronic characteristics produced by adding tiny quantities of Platinum to a Fe-Ni composition encased in graphitic nano-carbon shells accounted for the remarkable catalytic behavior and stability.¹³⁸ Manas Kumar Kundu and coworkers reported hybrids of Pt nanostructures and N-doped carbon as effective catalysts for the synthesis of hydrogen. The Pt-(PtO_x)-NSs/C catalyst showed 4 to 6.5 times greater HER activity during basic and acidic environments than commercial Pt/C. It reached 10 mA cm⁻² of current density in 0.5 M KOH at 51 mV of overpotential and 64 mV dec⁻¹ of Tafel slope. In acidic conditions, the HER activity of the Pt-(PtO_x)-NSs/C and PtO_x-free Pt Nanostructures/C (PtNSs/C) catalysts was equal. PtO_x increased the HER/HOR activity of Pt-(PtO_x)-NSs/C in an alkaline medium. In particular, the Pt-(PtO_x)-NSs/C catalyst had five times the activity of PtNSs/C under basic circumstances and equivalent activity under acidic conditions. As active sites for OH⁻ adsorption in basic environments, PtO_x helped build OH_{ads}, which subsequently reacted with nearby Pt sites' hydrogen intermediaries (H_{ads}) to produce H₂O and increase HER activity.¹³⁹ The Pt-doped HER electrocatalysts are summarised in Table 8.

5. Non-metallic heteroatom doping

The incorporation of non-metallic dopants into catalyst matrices offers an effective strategy for modulating catalytic behavior without altering the metallic lattice structure. Non-metals such as nitrogen, phosphorus, sulfur, and boron influence catalytic activity primarily by tuning the electronic properties of the active sites and inducing surface defects or

heteroatom functional groups.¹⁴³ Unlike metallic dopants that often form solid solutions, non-metallic elements typically bond covalently with the host, leading to charge redistribution and local distortion in the coordination environment. These effects can significantly impact key parameters like the hydrogen ΔG_{H^*} , which determines HER efficiency. Non-metallic doping can either donate or withdraw electrons, thereby shifting the electronic density at catalytic sites and influencing adsorption energetics. Although the volcano plot framework remains a useful tool to evaluate ΔG_{H^*} , many non-metal-doped systems show enhanced HER activity even when ΔG_{H^*} is sub-optimal. This suggests that additional factors such as improved electrical conductivity, increased defect density, and enhanced hydrophilicity also contribute to catalytic performance. For instance, nitrogen doping introduces lone-pair electrons that facilitate electron transfer, while sulfur doping can create favorable active edge sites. Therefore, when evaluating the role played by non-metallic dopants in HER catalysts, it is crucial to consider not just hydrogen binding energy but also the structural and electronic effects that alter the reaction pathway. These multifaceted enhancements will be further exemplified in the following sections.^{53,144,145}

5.1 P-doped electrocatalysts for HER

Phosphorus (P) provides more of an impact on the control of surface charge states because of its bigger atomic radius and enhanced electron-donating ability. Although introducing P from the same group to non-metallic elements found in pristine catalysts increases their catalytic efficiency, adding P from a different group to non-metallic atoms in hosting catalysts is expected to result in larger benefits. P enhances catalytic capabilities by influencing the electronic configurations of metal components. When P introduces itself, non-noble metal-based materials must be treated using phosphidation processes which reduce oxide impurities on the surface by reacting with active phosphorus compounds. P loading additionally changes the valence states of metal oxides, stabilizing oxygen vacancies, essential active sites for HER, and improving electrocatalytic activity. P doping boosts intrinsic activity toward the HER by increasing the electrical conductivity and optimizing hydrogen adsorption-free energy, as demonstrated by theoretical calculations and experimental investigations.¹⁴⁶⁻¹⁴⁹

Table 8 Pt-doped electrocatalysts for HER

S. no	Catalyst	Electrolyte	Overpotential (η) at 10 mA cm ⁻²	Tafel slope (mV dec ⁻¹)	C _{dl} (mF cm ⁻²)	Stability (hours & cycles)	Ref.
1	Pt@MTO-S	0.5 M H ₂ SO ₄	73	28	72.7 μ F	12 h	130
2	Pt@MoS ₂	—	88	30.24	32.68	1000 CV cycles	131
3	Pt/Co ₂ P/Ni ₂ P/NF	1.0 M KOH	75	64	30.4	25 h	132
4	Pt@DNA	0.5 M H ₂ SO ₄	26	30	—	24 h	133
5	Pt _{SA} -PtCo NCs/N-CNTs	1.0 M KOH	47	41	30	8000 CV cycles	134
6	MoS ₂ @Pt/YC	0.5 M H ₂ SO ₄	118	74	19.07	24 h	135
7	N-Pt-MoS ₂	1.0 M KOH	38	39	125	240 h	136
8	PtNPs/PEDOT _{0.3} SG _{0.7}	0.5 M H ₂ SO ₄	40	28.4	—	500 CV cycles	137
9	Pt-FeNi@C	0.5 M H ₂ SO ₄	50	26.1	5.6	10 h	138
10	Pt-(PtO _x)-NSs/C	0.5 M KOH	64	51	—	40 h	139



Suitable electrocatalysts for hydrogen production have been developed by Luozhen Bian and co-workers using Phosphorus (P)-doped MoS₂ nanosheets anchored on carbon cloths (CC). Molybdenum disulfide (MoS₂) nanosheets made on CC were directly loaded with P using a simple low-temperature phosphatization method that they developed. This resulted in a combined electrode for an effective hydrogen evolution process. Higher current densities for useful applications could be generated more easily with this method, which made it possible to fabricate phosphorus-doped MoS₂ on a flexible material. Increased internal basal plane active regions and improved material conductance were two ways that P stimulants in the MoS₂ matrix dramatically boosted HER performance. To reach current densities of 20 mA cm⁻² and 100 mA cm⁻², respectively, the optimized P-loaded MoS₂ nanosheets, which contained 3.3 at% phosphorus, exhibited much reduced overpotentials of 133 mV and 189 mV.

Additionally, they exhibited an outstanding tiny Tafel slope of 67.0 mV dec⁻¹.¹⁵⁰ Phosphorus-incorporated NiCo₂S₄ nanoparticles grown on carbon nanotubes that are embedded within a network of carbon nanofibers (P-NiCo₂S₄@CNT/CNF) are the basis of highly active electrocatalysts developed by Huahao Gu and co-workers, shown in Fig. 13a. CNTs were added using an easy co-electrospinning technique to improve the electrical insulation of the 3-D CNF system and to facilitate electron transmission to the connected HER active substance. With the help of this nanofiber network the electroactive NiCo₂S₄ developed onto the CNT/CNF template parallel by hydrothermal reaction; however, revealing additional catalytically active regions. Through phosphidation responses phosphorus loading into the hybrid enhanced the electrical composition of electroactive NiCo₂S₄, thereby lowering the energy resistance through the HER procedure. NiCo₂S₄'s electronic makeup was adjusted by loading phosphorus, a non-chalcogen heteroatom, to maximize the hybrid catalyst's hydrogen absorption energy. Lowering the kinetic energy barrier in the phosphidation process makes it easier to dope the hybrid with phosphorus. The P-NiCo₂S₄@CNT/CNF hybrid demonstrated exceptional HER performance, with an exceptionally low onset overpotential of 27 mV, an amazing current density of 10 mA cm⁻² at η as low as 74 mV, a staggering exchange current density of 0.79 mA cm⁻² and excellent long-term durability. This hybrid was made possible by the synergistic effects resulting from electrical improvement and nanostructured morphological and phosphorus doping-induced efficiency of the electronic framework.¹⁵¹ Haijing Yan and co-workers developed Phosphorus-enhanced tungsten nitride (W₃N₂)/reduced graphene oxide (P-WN/r-GO) as an electrocatalyst for the HER process that is additionally highly efficient and affordable. When phosphorus was added to r-GO and WN, the work function increased significantly and approached that of platinum metal. The peculiar characteristics of phosphorus, which include lone-pair electrons in 3p orbitals and unoccupied 3d orbital spaces, allow it to modify the surface charging state to generate local densities of charge, which is responsible for the amazing activity of phosphides. By modifying Phosphorus on

tungsten nitride in a sensible way one may increase the activity of tungsten nitride catalysts in order. This is possible by making use of P's important function in the HER and its impact on the electrical state of metal components, which in turn has a major effect on catalytic characteristics. In acidic conditions, the P-WN/rGO catalyst had a low onset overpotential of 46 mV, a Tafel slope of 54 mV dec⁻¹, and a significant exchange current density of 0.35 mA cm⁻². To attain a current density of 10 mA cm⁻² the catalyst required an overpotential of 85 mV.¹⁵² Zhangping Shi and co-workers offer a strategy for precisely adjusting the electrical composition of nanostructured Mo₂C using phosphorus testing, which results in a high efficiency that is on par with noble-metal-based electrodes in the HER process. Phosphorus provided a fascinating effect on Mo₂C by increasing the electron concentration close to the Fermi limit of Mo₂C, which reduced the Mo-H interaction and accelerated HER rates. The phosphorus added to pristine Mo₂C caused a gradual rise in the negative hydrogen-binding energy balance (ΔG_{H^+}), which was attributed to steric interference and electron migration by the phosphorus on the Mo₂C substrate. This effectively weakened the Mo-H connection. These findings were further supported by density functional theory computations. The presence of P-Mo bonds in P-Mo₂C@C and the direct interaction between dopant and carbide was consistent with XPS findings. The ensuing P-Mo₂C@C nanowires showed outstanding behaviour, with a small overpotential of 89 mV at a current density of 10 mA cm⁻² and amazing kinetic metrics with a Tafel slope of 42 mV dec⁻¹.¹⁵³ Jingyan Zhang and co-workers offered a straightforward, one-step hydrothermal technique for producing Phosphorus (P)-doped cobalt disulfide materials. By lowering the intrinsic energy for hydrogen atomic adsorption (ΔG_{H^+}) at the cobalt sites, phosphorus substances may be able to increase the electrocatalytic function of hydrogen evolution. Additionally, the metallic P-added CoS₂ improved HER function by speeding up HER rates and facilitating electron transport. It is conceivable for the P atom to move into the sulfide, replacing the S atom vacancy to form new bonds with other substances and cause lattice deformation, given the near-atomic distance and electronegativity of P to S. Having an overpotential of just 46 mV and a current density of 10 mA cm⁻² in a 0.5 M H₂SO₄ solution, the optimized P-doped CoS₂ catalyst in this work exhibited outstanding behaviour, shown in Fig. 13b.¹⁵⁴ P-doped MoO_{3-x} nanosheets were developed by Ling Li and co-workers as reliable and efficient electrodes for the evolution of hydrogen. They showed an overpotential of 166 mV in 0.5 M H₂SO₄ to get an average current density of 10 mA cm⁻². Additionally, they exhibited strong catalytic durability across a range of pH-varying electrolytes. The combined action of vacancies in oxygen and phosphorus element enrichment resulted in enhanced catalytic efficiency and durability. Phosphorus's weak electronegativity might make it easier for protons to adsorb and desorb, which would increase the catalyst's effectiveness for HER. Significant characteristics of P-added catalysts with plenty of oxygen vacancies included their fast HER kinetics, minimal HER overpotential, and



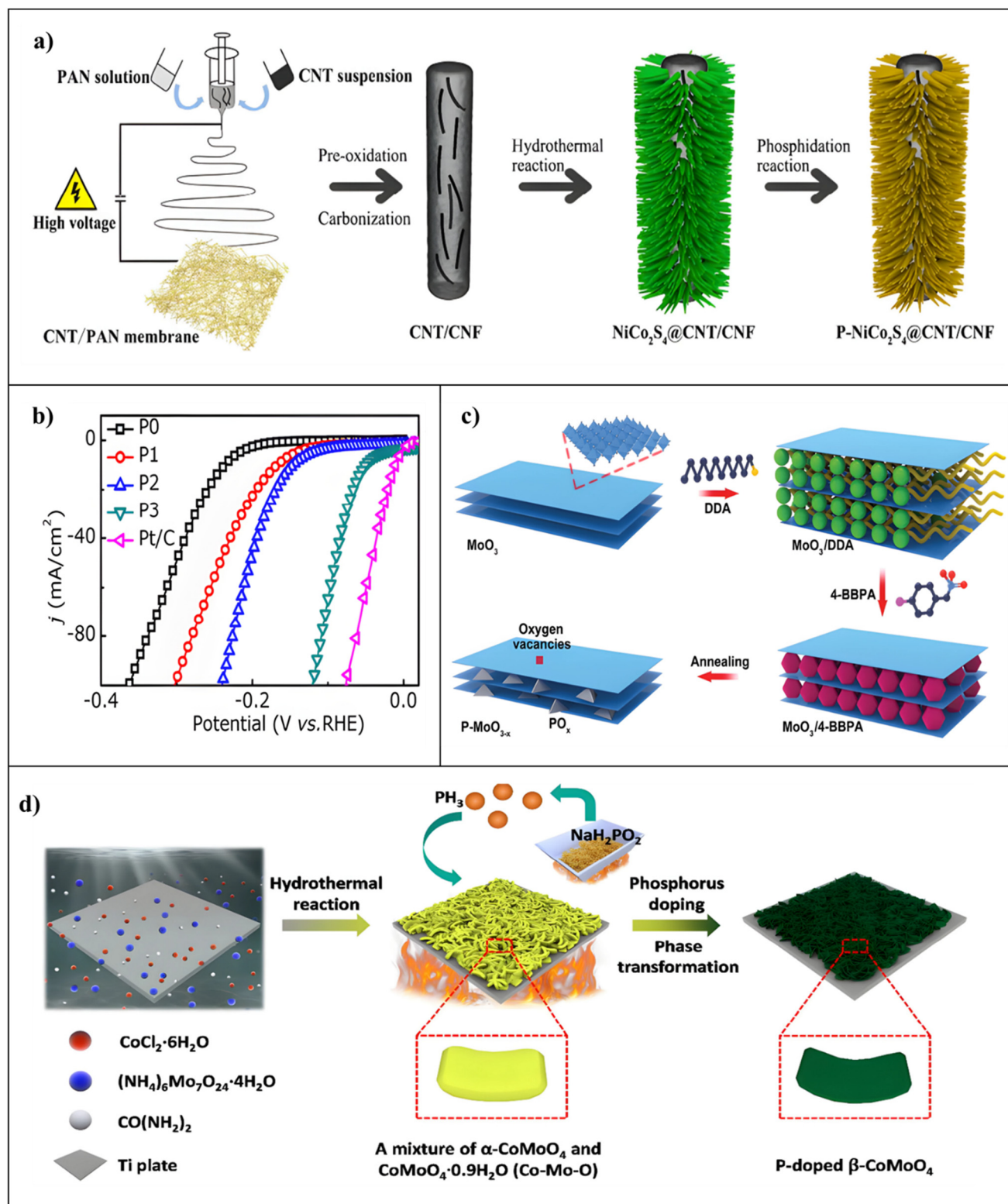


Fig. 13 (a) Diagrammatic representation of the P-NiCo₂S₄@CNT/CNF hybrid preparation process. Reproduced from ref. 151 with permission. Copyright 2017, RSC. (b) P-doped CoS₂ and pure CoS₂ polarisation curves in 0.5 M H₂SO₄ at a scan rate of 2 mV s⁻¹. Reproduced from ref. 154 with permission. Copyright 2018, Elsevier. (c) Diagrammatic representation of the P-MoO_{3-x} nanosheet preparation procedure. Reproduced from ref. 155 with permission. Copyright 2017, Wiley. (d) Diagrammatic representation of the P-doped β-CoMoO₄ synthesis process on a Ti plate. Reproduced from ref. 156 with permission. Copyright 2018, ACS.



superior stability in basic and acidic conditions, shown in Fig. 13c. The P-loaded MoO_{3-x} nanosheets that were produced exhibited notable hydrogen evolution resistance (HER) qualities, such as minimal overpotential and rapid transport of electrons.¹⁵⁵ Shu Li and co-workers showed that phosphorus enrichment of $\beta\text{-CoMoO}_4$ nanoplates transforms them into active HER electrodes. P-incorporated $\beta\text{-CoMoO}_4$ was incorporated directly onto the titanium plate to generate the self-supported electrode, which increased the number of active regions and improved electrolyte mobility through them. The electronic composition of $\beta\text{-CoMoO}_4$ was modulated by phosphorus doping, which also lowered the energy limit for H atom binding and increased its electrical mobility, shown in Fig. 13d. The enhanced electrical conductivity of $\beta\text{-CoMoO}_4$ following P doping was verified by both theoretical simulations and experimental findings. The hydrogen adsorption-free energy was shifted into an ideal area *via* phosphorus loading. The compositional alterations of P-doped βCoMoO_4 were studied using XPS. DFT simulations were used to investigate the impact of P doping on the HER performance of βCoMoO_4 . Such P-doped $\beta\text{-CoMoO}_4$ exhibited considerably improved HER catalytic activity when compared to pristine $\beta\text{-CoMoO}_4$, with an overpotential of only 138 mV needed to drive the formation of hydrogen at a current density of 10 mA cm^{-2} .¹⁵⁶ Ji-Sen Li and co-workers reported synthesizing MoP nanostructures with a metal-organic lattice assist technique and confining them within P-doped porous carbon (Mo-P@PC). The pore constriction effect of MOFs was responsible for the small size of Mo-P particles as its pore size distribution was calculated using DFT. XPS was used to figure out the surface valence and material composition of MoP@PC. Additionally, Mo-P nanomaterials were contained inside the carbon framework by the naturally occurring ligands of MOFs, which offered many exposed places for activity while preventing the particles from separating or accumulating. The carbon substrate also improved conductivity and decreased interfacial resistance. Phosphorus enrichment was essential for controlling the electrical structure of carbon atoms, which increased the porous carbon's conductivity considerably. With a Tafel slope of 59.3 mV dec^{-1} , an initial overpotential of 97 mV and exceptional long-term durability, the resultant Mo-P@PC composite demonstrated outstanding HER catalytic activity.¹⁵⁷ An approach for the rational development and easy construction of a MoS_2 -based dual-purpose electrocatalyst on carbon cloth (CC) was presented by Yan Hu and co-workers. Minimal overpotentials for hydrogen evolution were shown by the optimized P-doped MoS_2 @CoP/CC catalyst, with overpotentials of 64 mV in an alkaline solution and 72 mV in H_2SO_4 at a current density of 10 mA cm^{-2} . A thin layer of P-doped MoS_2 was formed on the Co-P nanoneedles' interface after arrangement of P-doped MoS_2 ; when the amount of the molybdenum precursor rose, greater coverage of the surface was observed. It is intriguing that, in the absence of a Co(OH)F buffer layer, the surface of P- MoS_2 modified CC showed significant clustering. XPS analysis of the charge transfer between Mo and Co spaces in the P- MoS_2 @CoP hybrid structure indicated synergy that

enhanced water dissociation, which contributed to faster kinetics. Nonetheless, an efficient charge transfer channel and open pathways for the quick release of gas bubbles after HER were provided by the nano-array framework formed *in situ* on the conducting CC.¹⁵⁸ A unique synthesis process for uniformly scattered NiCo-NiCoP nanoparticles implemented into P-doped CNTs (NiCo-NiCoP@PCT) was developed by Ziqian Zhou and co-workers. This method works by binding phytic acid (PA) on the outer layer of Ni-Co-precursor nanowires and then lowering the temperature. Because of their strong electrostatic attraction phosphorus atoms may draw electrons from metal atoms to reveal extremely active locations, and because of a negative charge they can also function as proton trapping, which gives rise to the amazing features of the ensuing TMPs. The remarkable electrocatalytic function of NiCo-NiCoP@PCT/CC resulted from the combined actions of PCT, which promoted effective electron flow, and the heterostructural NiCo-NiCoP that, because of its distinct structure, had a high intrinsic catalytic ability and many active regions. The NiCo-NiCoP@PCT, being a bi-functional electrode, had exceptional electrocatalytic performance for both hydrogen generation processes and achieved a minimal overpotential of 135 mV at 10 mA cm^{-2} .¹⁵⁹ The P-doped HER electrocatalysts are summarised in Table 9.

5.2 N-doped electrocatalysts for HER

The nitrogen (N) atom has three single electrons and one lone pair in its valence electron layer, providing N with a variety of bonding prospects. The tiny N atomic radius permits it to occupy intermediate locations in catalysts with ease while maintaining overall electrical conductivity.¹⁷⁰ Furthermore, N easily polarizes neighboring atoms due to its strong electronegativity, which affects catalytic activity.¹⁷¹ Ammonization is one way to add N atoms to host catalysts; this can be done by solid reactions with nitrogenous reagents such as urea or thiourea or by treating NH_3 through elevated temperatures. N-doping enhances conductive properties and helps charge transfer by maintaining the metallic nature of the host catalysts.¹⁷² In addition, N-doping can alter surface crystal arrangements, produce more active sites and reduce hydrogen adsorption pressures to enhance catalytic activity by improving HER reaction rates. In general, N atoms' empty orbitals provide advantageous conditions for Tafel and Volmer reactions, which encourages effective hydrogen production. Therefore, N-doping is a very beneficial approach for raising HER efficiency in catalysts based on non-noble metals.¹⁷³

To promote the evolution of hydrogen, Pengzuo Chen and co-workers designed Nitrogen Altered Metallic Co-S₂ Porous Nanowire Arrays, shown in Fig. 14a. Their method utilized a simple nitrogen designing process that made it easier to alter catalytic active sites, electronic arrangement, and reaction kinetics in metallic Co-S₂ hollow nanowire arrays in a coordinated manner. As confirmed by XPS and XANES spectra, adding nitrogen to the Co-S₂ material changed its shape to add more active sites and enhanced its electrical ability, which sped up charge transfer during the hydrogen evolution process (HER). The density functional model simulations further verified that the N



Table 9 P-doped electrocatalysts for HER

S. no	Catalyst	Electrolyte	Overpotential (η) at 10 mA cm ⁻²	Tafel slope (mV dec ⁻¹)	C _{dl} (mF cm ⁻²)	Stability (hours & cycles)	Ref.
1	P-doped MoS ₂	0.5 M H ₂ SO ₄	133 at (20 mA cm ⁻²)	67	17.6	1 h	150
2	P-NiCo ₂ S ₄ @CNT/CNF	0.5 M H ₂ SO ₄	74	65.9	33.8	10 h	151
3	P-WN/rGO	0.5 M H ₂ SO ₄	46	54	32	20 h	152
4	P-Mo ₂ C@C	0.5 M H ₂ SO ₄	89	42	15.6	3000 CV cycles	153
5	P-doped CoS ₂	0.5 M H ₂ SO ₄	53	57	49	10 000 CV cycles	154
6	P-MoO _{3-x} - <i>n</i>	0.5 M H ₂ SO ₄	82	42	—	1000 CV cycles	155
7	P-β-CoMoO ₄ /Ti	1.0 M KOH	138	68.76	—	20 h	156
8	MoP@PC	0.5 M H ₂ SO ₄	97	59.3	39.6	1000 CV cycles	157
9	P-MoS ₂ @CoP/CC	0.5 M H ₂ SO ₄	72	51.22	143	30 h	158
10	NiCo-NiCoP@PCT	1.0 M KOH	135	77.79	5.07	10 h	159

inclusion improved HER kinetics by optimizing the free energy of hydrogen adsorption for both sulphur and cobalt active sites. The resultant N-Co-S₂ NWS/CC electrode had a notably lower overpotential of 152 mV and a current density of 50 mA cm⁻², demonstrating a dramatically better HER performance.¹⁶⁰ To achieve extremely effective evolving hydrogen catalytic activity, Yishang Wu and co-workers developed NiCo₂-S₄ nanowires using N-inclusion. Because of the strong chemical bonds between adsorbed H and S, which is extremely electronegative, conventional metal sulphides employed for this purpose frequently have difficulties with hydrogen release. N-doping was repeatedly found to be an effective way to modify the exterior electron densities of NiCo₂-S₄ by a combination of XPS, synchrotron-based X-ray absorption spectral analysis, and density functional analysis tests. The strategic development of extremely effective catalysts for hydrogen evolution appears to be promising given the capacity to control electron concentrations at catalytic locations. XPS and DFT simulations revealed that N dopants may influence the electrical characteristics of NiCo₂S₄, enabling H* desorption from sulfur sites and HO-H cleavage in water molecules. The NiCo₂-S₄ nanowire networks that were nitrogen incorporated showed a Tafel slope of 37 mV dec⁻¹ and an overpotential of 41 mV at 10 mA cm⁻², shown in Fig. 14e.¹⁶¹ For the electrochemical hydrogen generation process, Junqing Yan and co-workers developed a single-step, consistent technique for producing Nitrogen-incorporated molybdenum-dioxide nanosheets. This entailed producing N-added molybdenum oxide materials by calcining urea and layered molybdenum oxide at extremely high temperatures. N increased the number of proton adsorption sites with lower binding energies and caused surface structure instability. The Surface N doping, surface bending, and the inherent characteristics of molecular oxide worked together to produce a considerable and long-lasting hydrogen evolution action, with a Tafel slope of 33 mV dec⁻¹ and an overpotential of 96 mV at 10 mA cm⁻².¹⁶² Ultrathin carbon nanosheets treated with N, initially developed by Hao Jiang and co-workers, provide a state-of-the-art metal-free electrocatalyst for the hydrogen-evolving process. They generated N-doped ultrathin carbon nanosheets (NCNs) *via* straightforward pyrolysis using a unique gas-foaming method, shown in Fig. 14b. The carbon atoms at the armchair border and next to the graphitic N dopants were determined by the team to be

inherent active sites for HER by DFT calculations. In acidic conditions, the catalyst also showed notable activity and durability for HER. The low Tafel slope of 43 mV dec⁻¹ of NCN confirmed its exceptional kinetic characteristics.¹⁶³ A new electrocatalyst for the hydrogen evolution process was presented by Chun Cheng Yang and co-workers. It is made of Fe₃C-Co nanoparticles enclosed in an N-doped carbon stacked framework (Fe₃C-Co/NC). To further understand the electrocatalytic activity of Fe₃C-Co/NC, the surface electronic state and chemical composition were measured by XPS. DFT calculations were performed to determine the mechanism of the Fe₃C-Co/NC hybrid's ORR and HER electrocatalytic performance. This unique hybrid material, which was manufactured for the first time by a simple template-removal technique, blends 2D carbon particles with 1D CNTs to produce many active sites from Fe₃-C, Co, and NC species; in the mesoporous carbon arrangement, Fe₃-C, Co, and NC are intimately linked. In both acidic and alkaline conditions the catalyst demonstrated strong HER activity; in acidic solutions the overpotential was 298 mV, and in alkaline solutions the overpotential was 238 mV.¹⁶⁴ For the hydrogen evolution process, Yana Men and co-workers designed N-doped CoP as a long-lasting electrocatalyst. The development of N-doped CoP nanoflowers that exhibited exceptional HER efficacy and were produced on conductive carbon cloth (N-N-doped CoP/CC) was covered. According to experimental findings, N-doping effectively modified the electrical properties of Co d-orbitals and maximized the value of the free energy of hydrogen absorption, leading to improve HER performance. N-loading enhanced the free energy of adsorbed H for CoP to an ideal level, according to DFT calculations, giving N-CoP/CC strong HER performance. With overpotentials of 39 mV and 25 mV, respectively, to reach a current density of 10 mA cm⁻² in 1.0 M KOH and 0.5 M H₂SO₄, the resultant N-doped CoP/CC nanoflowers showed impressive HER activity. The exceptional HER function and robustness of N-CoP/CC were attributed to its very open shape, greater exposure of the active regions, improved electrical properties, and simultaneous electronic effects induced by N doping. A broad pH range from 0 to 14 was demonstrated by N-doped CoP/CC's outstanding HER performance.¹⁶⁵ The impacts of doping transition metals (Fe, Ni, and Cu) on the electrocatalytic activity of M-Co₂P/N-doped-CNTs hybrid catalytic agents for the HER were studied by Yuan Pan and co-workers. The synthesis of



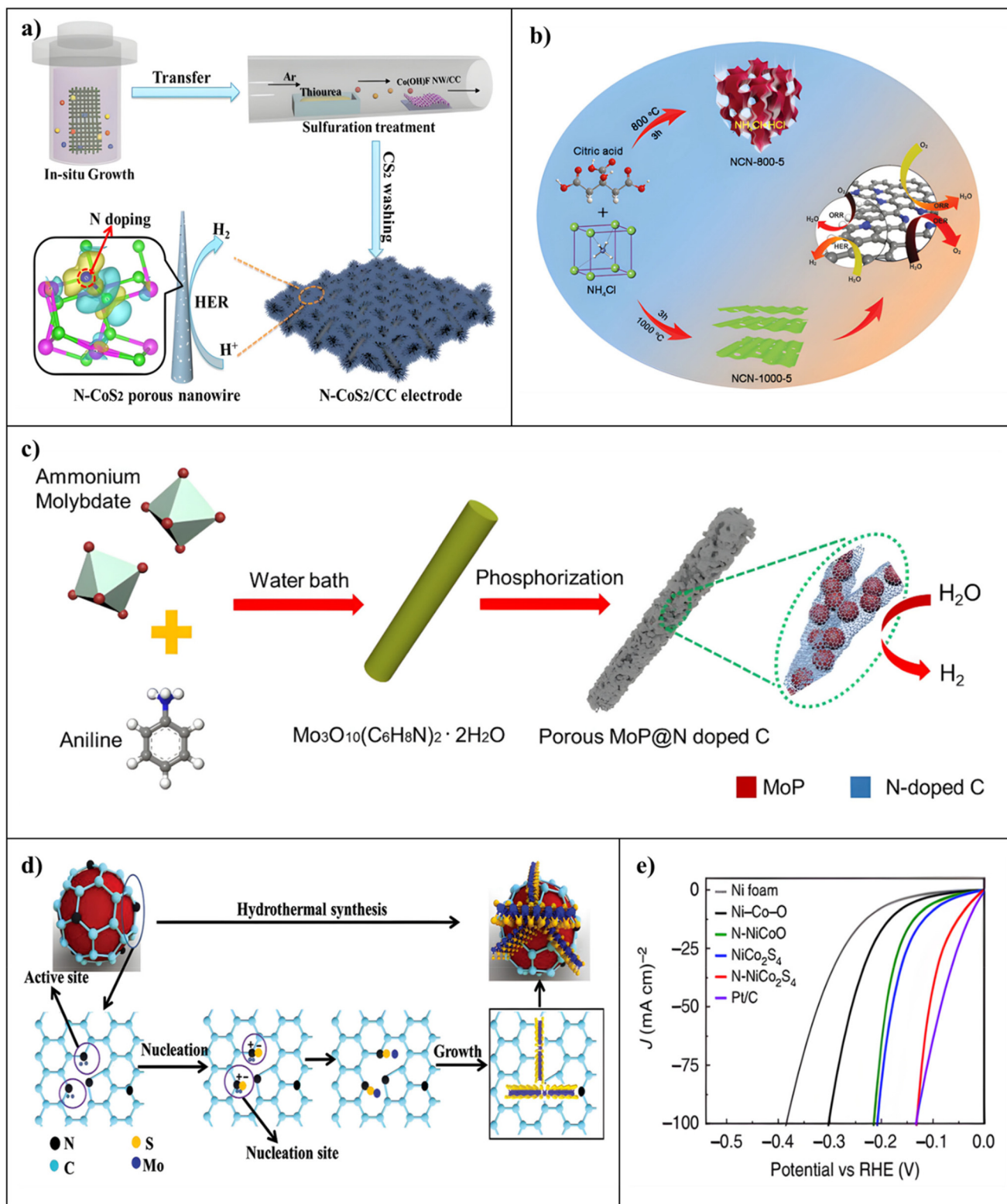


Fig. 14 (a) Illustration of the NCoS₂ NWs/CC electrode material synthesis process. Reproduced from ref. 160 with permission. Copyright 2017, ACS. (b) A schematic of the NCN synthesis process. Reproduced from ref. 163 with permission. Copyright 2019, RSC. (c) The porous MoP@NC hybrid NWs are shown schematically. Reproduced from ref. 167 with permission. Copyright 2019, Elsevier. (d) Diagram showing the process by which the Ni@NC@MoS₂ product forms. Reproduced from ref. 168 with permission. Copyright 2019, Wiley. (e) LSV curves in a 1.0 M KOH solution with a scan rate of 5 mV s⁻¹. Reproduced from ref. 161 with permission. Copyright 2018, Nature.



these hybrid catalysts was carried out by a simple *in situ* thermal degradation method. It was discovered that doping with Fe atoms improved the crystal plane's development, increasing its specific surface area and exposing more catalytically active spots. Furthermore, upon doping, Fe²⁺ showed the largest positive charge of all the M-Co₂P/N-doped-CNT hybrid catalysts. Fe-Co₂P/N-doped CNTs, Ni-Co₂P/N-doped CNTs, and Cu-Co₂P/N-doped CNTs had onset overpotentials of 25 mV, 96 mV, and 140 mV according to the comparison study.¹⁶⁶ N-doped carbon-plated highly porous Mo-P Nanowires have been developed by Chaoran Pi and co-workers as a very effective catalyst for the HER. They phosphorized organic-inorganic hybrid NWs to generate heterogeneous Mo-P@NC NWs, shown in Fig. 14c. The H⁺ utilization was facilitated by the *in situ* development of NC, which was characterized by many tiny pores. Mo-P@NC exhibited synergistic advantages, many active sites, and quick ion/electron transport. DFT calculations revealed the strong electrocatalytic activity of the MoP@NC hybrid electrocatalyst. The XPS results indicated that the bonds in MoP@NC remained constant after cycling. However, without the carbon layer, the Mo-P bond decreased significantly, confirming the high oxidation resistance of the carbon covering. Excellent HER function and endurance were demonstrated by the electrocatalyst throughout a broad pH range. In acidic and alkaline solutions, correspondingly, modest overpotentials of 96 mV and 149 mV were notable, along with Tafel slopes of 49.2 mV dec⁻¹ and 61.7 mV dec⁻¹. These results were supported by remarkable stability.¹⁶⁷ Ni@N-treated Carbon@MoS₂ Nanosheets have been proposed by Sayyar Ali Shah and co-workers as a very effective electrocatalyst for hydrogen evolution operation. Ni@NC@MoS₂ pyramidal microspheres were generated by vertically growing Mo-S₂ nanosheets with enlarged interfacial spacings on a Ni@N-doped carbon (Ni@NC) material using a straightforward hydrothermal approach, shown in Fig. 14d. The synergistic impact of compact catalytic regions on Mo-S₂ nanosheets with exposed margins and enlarged inter-layer distances, together with fast electron transport from the Ni@NC surface to Mo-S₂ nanosheets, was responsible for the excellent HER efficiency of the Ni@NC@Mo-S₂ catalysis. A very small Tafel slope of 47.5 mV dec⁻¹ and a modest overpotential of 82 mV at 10 mA cm⁻² indicated that this exceptional electrocatalyst has great potential for use in practical hydrogen generation operations.¹⁶⁸ Polyaniline-induced N-enriched carbon-

coated Co-P particles embedded in N-Infused Graphene is the effective electrocatalyst that Jingwen Ma and co-workers developed for the HER. N-added graphite nanosheets promoting N-doped carbon-coated CoP nanoparticles were generated in this production by pyrolysis and polyaniline-based phosphating methods. As the framework, N-doped graphene effectively inhibited the CoP nanoparticles from collecting, and the carbon shell made of polyaniline functioned as a shield, preventing the dissolution of the particles and improving their electrical mobility. In 0.5 M H₂SO₄, the electrocatalyst that was produced showed remarkable electrochemical activity for HER, exhibiting a low Tafel slope of 59.3 mV dec⁻¹ and a modest overpotential of 135 mV at 10 mA cm⁻². Moreover, after 14 h of operation, the N-doped carbon shells enclosing the CoP nanoparticles demonstrated outstanding durability and resistance to corrosion.¹⁶⁹ The N-doped HER electrocatalysts are summarised in Table 10.

5.3 B-doped electrocatalysts for HER

Boron has three valence electrons and an electron deficiency; it is more likely to bond with oxygen single-electron pairs from H₂O molecules, which facilitates the release of protons during the HER catalysis process.¹⁸³ The redistribution of electrons made possible by B-doping strengthens the crystal structure of catalyst substances, lowering the possibility of corrosion or structural degradation over extended HER operation.¹⁸⁴ Furthermore, the HER's reaction kinetics may be changed by boron dopants, resulting in reduced overpotential and higher reaction rates. Thus, the catalyst's total energy efficiency is raised as a consequence of this kinetic increase, which leads to more productive hydrogen synthesis at lower energy sources. Boron dopants are a useful tactic for accelerating the catalytic activity of catalysts based on non-noble metals. Nevertheless, there are currently few publications on B-doped non-noble metal-based catalysts, which points to a unique possibility for the synthesis of innovative catalysts using B-doping.

By synthesizing metal-liberated B-loaded graphene, Bhaskar R. Sathe and co-workers demonstrated exceptional electrocatalytic ability in accelerating the hydrogen evolution process. Reduced conversion barriers were ascribed to the faster conversion of H⁺ ions to H₂, which was made possible by the integration of B heteroatoms as well as by C-B relationships, within the graphene framework, shown in Fig. 15a. The

Table 10 N-doped electrocatalysts for HER

S. no	Catalyst	Electrolyte	Overpotential (η) at 10 mA cm ⁻²	Tafel slope (mV dec ⁻¹)	C _{dl} (mF cm ⁻²)	Stability (hours & cycles)	Ref.
1	N-CoS ₂ NWs/CC	0.5 M H ₂ SO ₄	152 at (150 mA cm ⁻²)	58	—	3000 CV cycles	160
2	N-NiCo ₂ S ₄	1.0 M KOH	41	37	18	1000 CV cycles	161
3	N-MoO ₂	0.5 M H ₂ SO ₄	96	33	3	1000 CV cycles	162
4	NCN-1000-5	0.1 M KOH	51	43	—	500 CV cycles	163
5	Fe ₃ C-Co/NC	1.0 M KOH	238	108.8	0.75	16.6 h	164
6	N-CoP/CC	1.0 M KOH	39	58	—	30 h	165
7	M-Co ₂ P/NCNTs	0.5 H ₂ SO ₄	104	68	15.4	1000 CV cycles	166
8	MoP@NC NW	1.0 M KOH	149	61.7	14.15	5000 CV cycles	167
9	Ni@NC@MoS ₂	0.5 H ₂ SO ₄	82	47.5	4.35	3000 CV cycles	168
10	CoP@NC-NG	0.5 H ₂ SO ₄	135	59.3	—	14 h	169



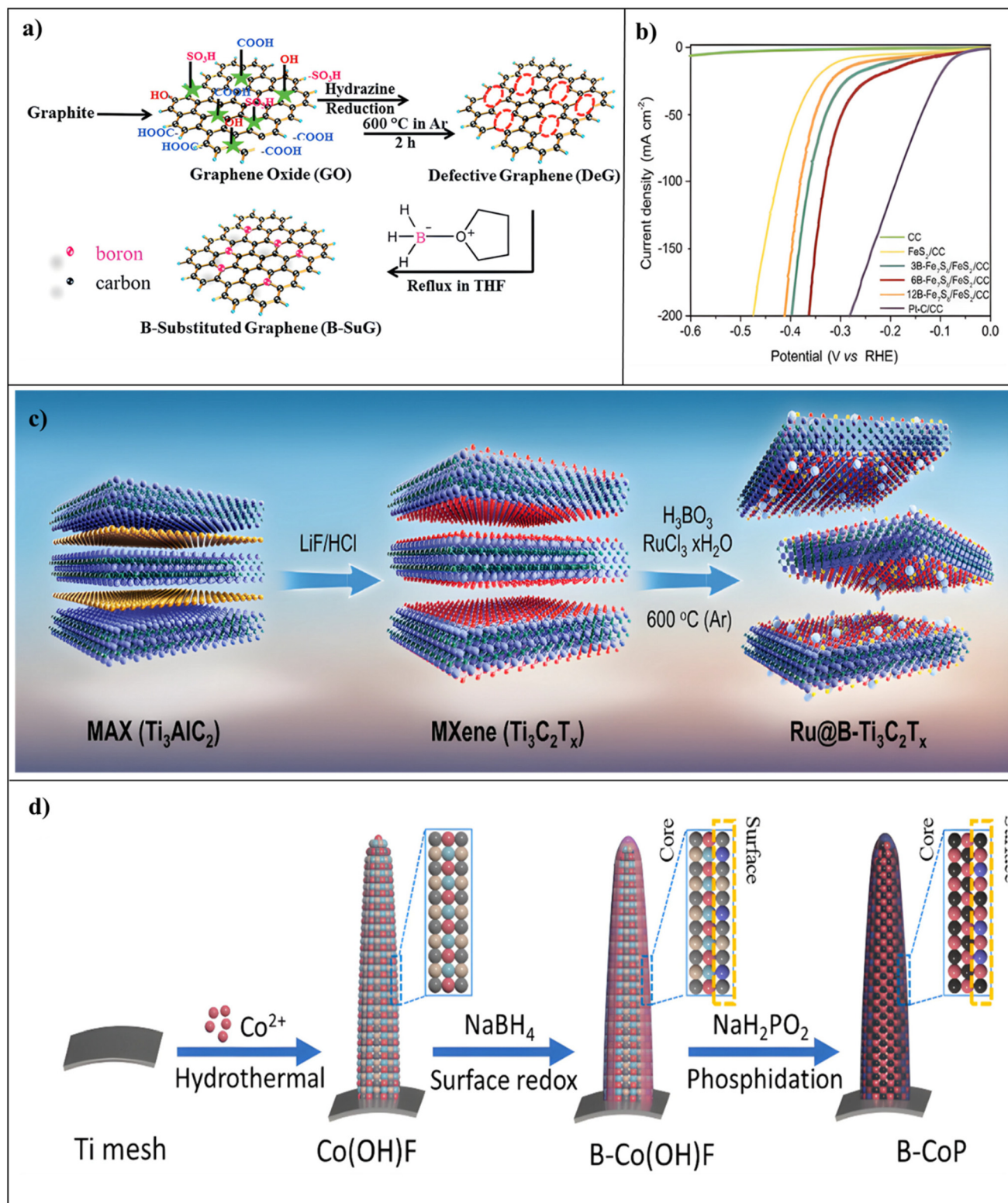


Fig. 15 (a) Schematic representation of producing B-substituted graphene (B-SuG). Reproduced from ref. 174 with permission. Copyright 2018, RSC. (b) The B-Fe₇S₈/FeS₂/CC electrocatalysts HER performance in a 1 M KOH solution. Reproduced from ref. 45 with permission. Copyright 2022, Wiley. (c) A schematic representation of the synthesis of Ru@B-Ti₃C₂T_x and Ti₃C₂T_x. Reproduced from ref. 178 with permission. Copyright 2021, Wiley. (d) Synthesis procedure of B-CoP. Reproduced from ref. 181 with permission. Copyright 2023, Elsevier.



unique material was produced by carefully replacing the carbon atoms in graphene that were inadequate with B atoms using borane tetrahydrofuran, a borylating chemical that is widely accessible. When it comes to borylating substances, $\text{BH}_3\text{-THF}$ is the best option since it produces the most effective electrocatalyst. When B-SuG was applied in the HER in a 0.5 M H_2SO_4 solution, there was a significant increase in current density at an overpotential as low as ~ 0.2 V vs. RHE.¹⁷⁴ To boost the alkaline HER, Jing Wu and co-workers developed B-doped $\text{Fe}_7\text{S}_8/\text{FeS}_2$ catalysts. DFT simulations were carried out to assess the intrinsic activity of the electrocatalysts. The Iron and Sulfur atoms' electronic arrangements were successfully tuned by adding B atoms to the $\text{Fe}_7\text{S}_8/\text{FeS}_2$ electrodes. By reducing energy barriers for splitting water and increasing the efficiency of desorbing H_2 intermediate molecules, this optimization resulted in the production of ideal B- $\text{Fe}_7\text{S}_8/\text{FeS}_2$ catalysts. XPS was used to examine the electronic structures, composition, and chemical states of B- $\text{Fe}_7\text{S}_8/\text{FeS}_2$ electrocatalysts. This HER electrocatalyst had a 113 mV overpotential in alkaline conditions, enabling a current density of 10 mA cm^{-2} , shown in Fig. 15b. Heteroatom adding shows promise in altering the electrical characteristics of transition metal sulfide electrocatalysts, hence speeding their kinetics in hydrogen evolution processes.⁴⁵ Boron-added Co- Se_2 nanowires were developed by Zhiming Liu and co-workers to serve as very effective catalysts for the hydrogen evolution process. The effective production of B-added Co- Se_2 nanoparticles on carbon cloth (boron-Co Se_2/CC) was shown by their study. XPS was employed to examine Co Se_2/CC and B-Co Se_2/CC to assess the impact of B doping on the electronic structure. Co- Se_2 has intrinsic catalytic properties; however, its effectiveness is limited by the poor hydrogen adsorption on its active cobalt site. Anions were added to the catalyst to increase its electronic framework, which in turn improved HER efficiency. Simulations using DFT demonstrated that B loading improved Co- Se_2 's intrinsic conductance and maximized the hydrogen ΔG_{H^*} at the selenium and cobalt sites. At 10 mA cm^{-2} , the boron-Co Se_2 catalyst produced an overpotential of just 153 mV, while the boron-Co Se_2/CC sample's Tafel slope was determined at a modest 85 mV dec^{-1} .¹⁷⁵ Incredibly robust Ultrafine Boron-treated NiCo@Carbon Nanocrystals were reported by Bashir Adegbeemiga Yusuf and co-workers as a robust catalyst for the HER. They developed extremely stable ultrafine NiCo@boron-treated carbon (NiCo@BC) hybrids using a simple reduction strategy. The remarkable catalytic activity resulted from the complementary interactions among the NiCo and B-treated carbon stages in the hybrid substance, which has a crystalline structure, many catalytically active areas, and good electrical properties. The partial transfer of electrons from B to NiCo metals was made easier by the abundance of active sites on the catalyst's surface, which greatly increased its catalytic efficiency. As such, this material exhibits great potential for use as an electrocatalyst for HER in alkaline environments. With a modest Tafel slope of 60 mV dec^{-1} and a low geometrical overpotential of 209 mV at a current density of 10 mA cm^{-2} , the produced NiCo@BC nanoparticles catalyst

demonstrated outstanding HER performance.¹⁷⁶ Qin Lin and co-workers developed boron-incorporated molybdenum carbide as an independent electrocatalyst for the hydrogen generation process. Boron doping was introduced, which decreased the free energy for H^* adsorption while increasing intrinsic reactivity. In the B-MoC framework, this alteration promoted charge movement and inhibited nanoparticle aggregation. Moreover, excess electron transfer to MoC nanoparticles caused by B-doping's induction of electron shortage resulted in the development of multiple proton adsorption sites and an increase in electrocatalytic activity. With outstanding stability and a low overpotential of 285 mV at a current density of 10 mA cm^{-2} , the B-MoC also performed well over a wide pH range.¹⁷⁷ Munkhjargal Bat-Erdene and co-workers reported that boron-capped $\text{Ti}_3\text{C}_2\text{T}_x$ (MXene) nanosheets are an approach to improve electrocatalytic hydrogen production in an additive way. In applications involving electrocatalytic Ru nanoparticle encapsulation, these heteroatom-doped $\text{Ti}_3\text{C}_2\text{T}_x$ (MXene) nanosheets served as effective solid substrates, shown in Fig. 15c. XPS revealed the effective synthesis of $\text{Ti}_3\text{C}_2\text{T}_x$ and its oxidative states. Electrochemical experiments and quantum structural first-principles simulations have shown that adding B onto 2D MXene nanosheets significantly improved the rate of intermediate H^* adsorption and lowered the charge-transfer barrier during the HER mechanism. It increased active site responsiveness, and advantageous electrode kinetics resulted from this improvement. Upon achieving current densities of 10 mA cm^{-2} and 100 mA cm^{-2} , respectively, for the HER, the Ru@B- $\text{Ti}_3\text{C}_2\text{T}_x$ nanosheets exhibited excellent catalytic activity and had tiny overpotentials of 62.9 mV and 276.9 mV.¹⁷⁸ A one-pot fabrication technique for boron-loaded C-enclosed Ni₃-Fe nanoparticles has been presented by Yusuf Bashir Adegbeemiga and co-workers, displaying excellent activity for hydrogen evolution processes. In alkaline environments, the Ni₃Fe@BC compound performed better electrochemically, suggesting a synergistic interaction between the Iron and Nickel components. The catalytic activity of Ni₃-Fe@BC-500 °C was attributed to the doping of B and the corresponding bonding of metal ions (Fe + Ni), which may augment the sites of catalysis. The one-pot production procedure of Ni₃-Fe@BC makes it extremely promising for HER electrocatalysis applications by utilizing low-cost component sources. Moreover, the high electrochemical surface area, boron-to-metal electron transfer, and tiny particle size were responsible for the remarkable electrochemical performance. Notably, Ni₃-Fe@BC exhibited tiny Tafel slopes, low overpotentials, and good stability for HER: it had a Tafel slope of 77 mV dec^{-1} and an overpotential of 280 mV.¹⁷⁹ The B-added amorphization of LDH was developed by Hongyuan Yang and co-workers to enable large current-density hydrogen evolution processes. As opposed to other materials where boride/phosphide manufacturing occurred from Boron/Phosphorus inclusion into LDHs for HER, this amorphization, caused by boron doping, enhanced HER activity. Furthermore, it is easy to use this adjustable B doping-induced amorphization technique with other transition metal (TM) LDHs. The remarkable effectiveness of



A-NiCo LDH/NF has been attributed to its distinct three-dimensional arrangement and the modulation of the crystal-line-to-amorphous framework, consequently increasing the capacity for charge transfer and revealing active sites. The amorphous boron-incorporated Ni-Co LDH supported on nickel foam (A-NiCo LDH/NF) exhibited remarkable robustness in alkaline conditions even after 72 h, and could achieve huge current densities at modest overpotentials of 1000 mA cm⁻² at 381 mV, 500 mA cm⁻² at 286 mV, and 100 mA cm⁻² at 151 mV.¹⁸⁰ Using controlled surface redox processes, Yingying Gao and co-workers reported a technique for boron surface layer restricted doping on cobalt phosphide nanowire arrays, shown in Fig. 15d. At 100 mA cm⁻², the resultant nanowire CoP with a B-doped surface layer composition was an outstanding electrocatalyst for the hydrogen evolution process, obtaining an overpotential of 112 mV. Cobalt is given electron-rich properties *via* boron doping, which causes electron rearrangement inside the material according to theoretical understanding and XANES and XPS investigations. These low-valence cobalt centres have a significant affinity for H₂O molecules and maximize Δ*G*_{H⁺}, which speeds up reaction kinetics when coordinated with boron. Cobalt centres doped with boron on the surface layer showed increased HER activity in comparison with non-doped CoP. B-doping causes electron redistribution and changes the d-band structure of cobalt which promotes the produce of low-valence cobalt centres and, in the end leads to optimized Δ*G*_{H⁺} and quicker reaction kinetics, as both experimental results and DFT simulations showed.¹⁸¹ A boron doping method was developed by Qingping Yu and co-workers with initial intense metal-support connections of ultra-small Rh particles embedded in an N-doped carbon framework (B-Rh@NC). It was confirmed by both theoretical and experimental research that boron doping maximized its electronic makeup and water adsorption power, hence triggering the metal-support connections between Rh and NC. As such, B-Rh@NC demonstrated excellent electrocatalytic activity for the HER over a wide pH range in addition to outstanding durability. To further increase HER activity boron loading efficiently controls the electrical framework and the absorption of water energy. Boron doping improves HER activity by regulating electrical structure and water adsorption energy, based on DFT analysis. On obtaining

low overpotentials in 1.0 M KOH (26 mV) and 0.5 M H₂SO₄ (43 mV), the resulting B-Rh@NC nanospheres exhibited activity for HER. Additionally, a current density of 10 mA cm⁻² in 1.0 M KOH could be generated by the built two-electrode system with just 1.53 V when B-Rh@NC was used as the cathode and Ru-O₂ was used as the anode.¹⁸² The B-doped HER electrocatalysts are summarised in Table 11.

5.4 O-doped electrocatalysts for HER

Oxygen has a greater electronegativity (3.5) than boron, which means it has more of an impact on the movement of electrons after doping. Higher conductivity and an extensive number of active sites are balanced by the introduction of oxygen heteroatoms. Determining the relationship between catalytic effectiveness and oxygen doping levels is essential. To facilitate effective electron transport during the HER, oxygen doping accelerates charge transfer kinetics at the catalyst-electrolyte interface. Increased performance optimization of the catalyst in the HER is possible by using oxygen doping to regulate surface morphology.¹⁹³ Reduction of band gap with oxygen incorporation, which improves intrinsic conductivity and facilitates proton-catalyst contact, is responsible for the increased catalytic activity according to DFT simulations.^{195,196}

Through an increasing phosphatization and dip-coating method, Yan Wen and co-workers developed O-doped hierarchical NiCoP/Ni₂P hybrid electrodes (O-NiCoP/Ni₂P) to reduce additional costs related to heteroatom doping. To achieve an effective alkaline HER, the resultant O-NiCoP/Ni₂P electrode showed an extremely low overpotential of 58 mV at 10 mA cm⁻² and retained long-term stability for 48 h. The HER kinetics increased by O doping, which maximized the free energy of hydrogen/water absorption, according to simulations using DFT simulations. Its catalytic efficacy was further enhanced by the strong coupling in O-NiCoP/Ni₂P that further improved electron transport. The transition-metal phosphide-based HER electrocatalysts reported by this work offer a viable method for their rational design.¹⁸⁵ The development of O-incorporated MoS₂ thin nanosheets for effective hydrogen evolution was developed by Junfeng Xie and co-workers. Several studies were made to improve the conductivity or the number of active sites in MoS₂ catalyst to increase the HER performance. The unsaturated S-atoms were made available as

Table 11 B-doped electrocatalysts for HER

S. no	Catalyst	Electrolyte	Overpotential (η) at 10 mA cm ⁻²	Tafel slope (mV dec ⁻¹)	<i>C</i> _{dl} (mF cm ⁻²)	Stability (hours & cycles)	Ref.
1	B-SuG	0.5 M H ₂ SO ₄	130	99	—	18 h	174
2	B-Fe ₇ S ₈ /FeS ₂	1.0 M KOH	113	57	12.4	1000 CV cycles	45
3	B-CoSe ₂ /CC	1.0 M KOH	153	85	58	24 h	175
4	NiCo@BC	1.0 M KOH	209	60	08.63	200 h	176
5	B-MoC	1.0 M KOH	285	128	17.28	1000 CV cycles	177
6	Ru@B-Ti ₃ C ₂ T _x	0.5 M H ₂ SO ₄	62	100	1.39	1000 CV cycles	178
7	Ni ₃ Fe@BC	1.0 M KOH	330	127	—	20 h	179
8	A-NiCo LDH/NF	1.0 M KOH	151 at (100 mA cm ⁻²)	94	12	72 h	180
9	B-CoP	1.0 M KOH	112 at (100 mA cm ⁻²)	61.9	—	20 h	181
10	B-Rh@NC	1.0 M KOH	26	52	18.6	10 h	182



HER targets by the formation of disorder, and the addition of oxygen efficiently modified the electronic configuration to increase intrinsic conductance, shown in Fig. 16a. An improved catalyst with a moderate amount of disorder was developed *via* controlled disorder design and oxygen incorporation, and it showed better electrocatalytic hydrogen evolution performance. HER activity is significantly increased when regulated disorder design and simultaneous oxygen inclusion in MoS₂ catalysts work together to regulate structural and electronic characteristics synergistically. DFT calculations showed that oxygen inclusion can govern the electronic structure of MoS₂, resulting in improved properties. To analyze the chemical composition of the products, XPS and elemental mapping studies were performed. The catalyst that has been tuned has an initial overpotential as low as 120 mV overall.¹⁸⁶ For the hydrogen evolution reaction, O-included WS₂ particles with improved electrocatalytic activities have been presented by Prasad V. Sarma and co-workers. Using chemical exfoliation along with ultrasonication processes, as well, they provide a unique way for the controlled generation of very catalytically active O-incorporated 1T and 2H WS₂ tiny clusters from O-deficient WO₃ tiny rods. The generated 1T tiny clusters had an extremely low Tafel slope of 47 mV dec⁻¹, minimal onset overpotential of 88 mV, large exchange current density, and superior stability. They were further distinguished by customized edge sites and improved conductivity, assigned to the metallic 1T phase and O-integration.¹⁸⁷ The synthesis of O-doped molybdenum disulfide nanosheets developed on carbon cloth, which produced a strong hydrogen generation process, was reported by Junfeng Xie and co-workers. On MoS₂, they together used elemental integration, disorder construction, and material bonding to enhance active sites and electrical conduction in a synergistic way that produced a very effective hydrogen-evolving efficiency. Frequently, the 2D electron coupling system can be harmed or degraded by the exposure of highly active edge regions of conducting MoS₂, which results in a decreased density of efficient active sites that are essential for activating the HER reaction. However, efficient synergistic improvement of both conductive properties and active sites was obtained by the combination of oxygen dopants with a distinct disordered shape. Consequently, the O-doped MoS₂/carbon fabric catalyst that was directed vertically exhibited an ultralow initial overpotential of 90 mV to start the HER activity.¹⁸⁸ O-doped cobalt/iron disulfides have been produced by Chaonan Wang and co-workers as highly successful catalysts for the generation of hydrogen process. While O-doped sulfur compounds were produced *via* a procedure encompassing calcination and sulfuration, Iron was added by exchange of ligands. Exceptional electrocatalytic HER performance was demonstrated by an optimized sample of CC@O-Co-Fe-S which showed remarkable endurance and an exceptionally low overpotential of 105 mV at a current density of 10 mA cm⁻². The shape of the O-Co-Fe-S nanosheet arrangement allowed for optimum exposure of the electrocatalytically active spots. O-adding greatly boosted the inherent catalytic activity and successfully

modified the electronic framework, causing the catalyst's charge to disperse. XPS was employed to examine the element compositions and electronic states of CC@O-CoFeS-0.025-500 and control samples. DFT studies showed that pure CoS₂'s Co atoms were catalytically active, with a Gibbs free energy of 0.38 eV for hydrogen adsorption (ΔG_{H^*}).¹⁸⁹ [W-O]-incorporated CoP was reported by Ge Meng and co-workers to improve hydrogen evolution at substantial current densities. A versatile catalyst for HER was produced when the [W-O] group, which is well known for its powerful adsorption ability, was incorporated into CoP. In addition to acting as strong H₂O adsorption locations which accelerates the cathodic H₂O splitting, these [W-O] species also aided in the Heyrovsky step by changing the d-band center of cobalt sites due to the connected O in the [W-O] group, which led to markedly improved catalytic results. Using this unique method, a hybrid catalyst with outstanding catalytic capabilities was produced by introducing the [W-O] grouping, which has a great adsorption capacity, into CoP nanoflakes. The theoretical computations were carried out using the DFT approach. XPS measurements were carried out to analyze the surface composition and binding structure of the catalysts. The catalyst achieved a minimal overpotential of 185.60 mV at 1000 mA cm⁻², shown in Fig. 16b.¹⁹⁰ A better electrocatalyst for the reaction of hydrogen evolution has been engineered by Jun Xu and co-workers in the form of O-doped VS₄ small particles with abundant sulfur vacant positions. The VS₄ spheres, which were produced using a simple solution process, have a significantly reduced band gap and enhanced intrinsic electrical conductivity due to their O-doping and abundance of S-vacancies. Fast charge transfer throughout the HER operation was made possible by this improvement. Furthermore, many highly active areas with high catalytic activity were provided for the proton-to-hydrogen conversion by the abundant defects of O-heteroatoms and S-vacancies. XPS was used for assessing the chemical states of the VS₄ and VS₂ samples. DFT studies showed that O-doping and S-vacancy significantly reduced the band gap of the VS₄, increasing its inherent electrical conductivity. As a HER catalyst, VS₄ nanoparticles outperformed 1T-VS₂ microflowers with an arrangement of layers owing to the combined benefits of defect design and electronics. With a modest Tafel slope of 44 mV dec⁻¹ and a low overpotential of 48 mV at 10 mA cm⁻², the VS₄ spheres catalyst exhibited impressive kinetic characteristics.¹⁹¹ For effective electrocatalytic water splitting, Thi Luu Luyen Doan and co-workers have developed a technique for producing oxygen-doped cobalt phosphide-layered shelled metal nanowires. Using a straightforward method they attached O-doped cobalt phosphide-layered shelled Cu nanowires vertically onto a 3D framework to build a hierarchical catalyst, shown in Fig. 16c. Significant flaws were introduced by this special construction, which also modified the surface science. The product had an expanded surface area with plenty of pathways for the diffusion process, rich electroactive sites, and regulated adsorption strength. Because of the small energy barrier for H adsorption, Co-based catalysts stand out among other alternatives due to their relative availability, affordability,



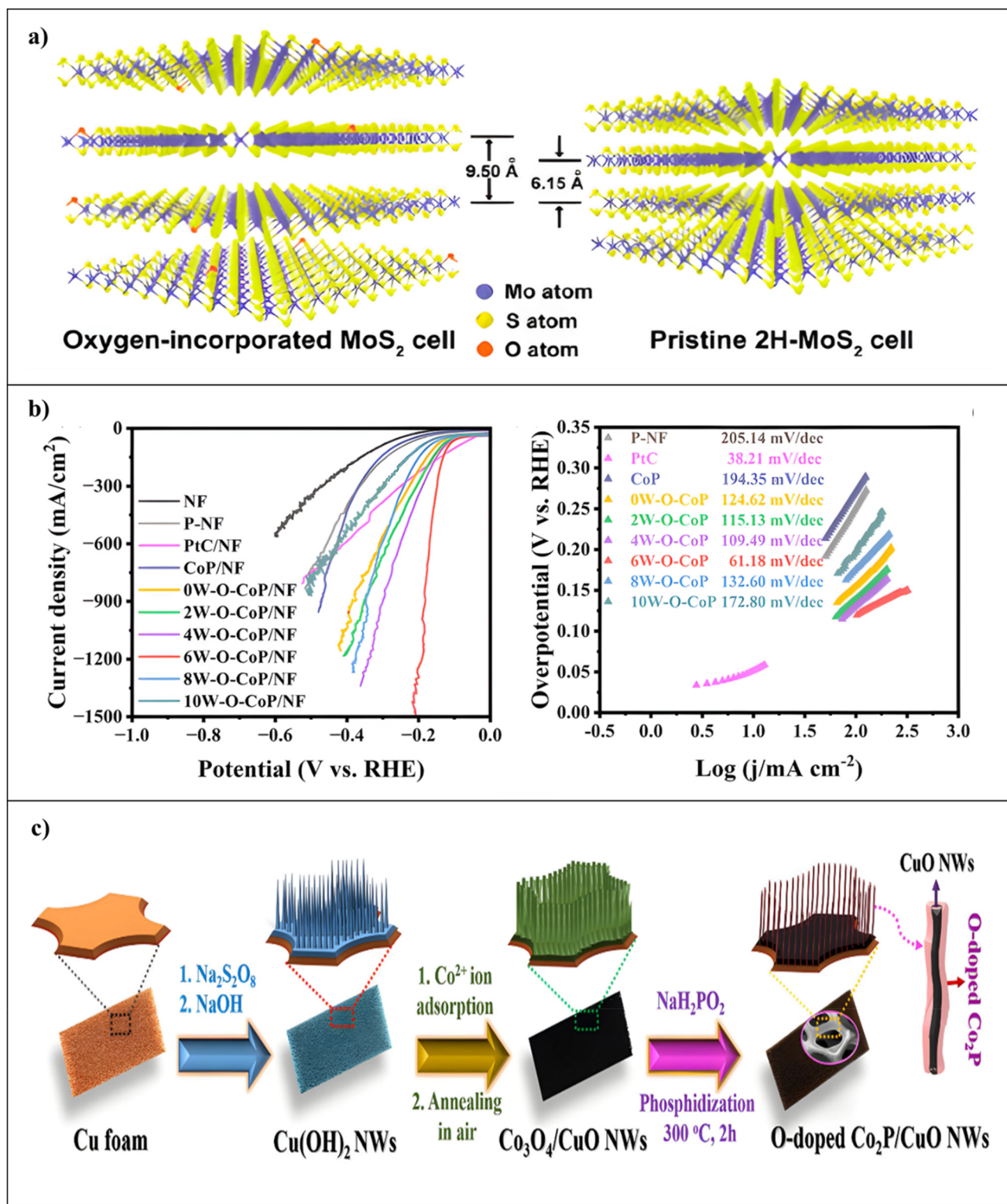


Fig. 16 (a) TGA and DTA curves of ultrathin MoS₂ nanosheets with oxygen incorporation. Reproduced from ref. 186 with permission. Copyright 2013, ACS. (b) HER polarisation curves for catalysts based on NF, P-NF, CoP/NF, commercial PtC/NF, and xW-O-CoP/NF (x = 0, 2, 4, 6, 8, 10) together with the matching Tafel slopes. Reproduced from ref. 190 with permission. Copyright 2023, Springer. (c) Diagram showing the steps involved in creating the O-doped Co₂P/CuO NWs/CF hybrid. Reproduced from ref. 192 with permission. Copyright 2019, Elsevier.



and high activity. For the HER process, the catalyst showed modest overpotentials of 101 mV resulting in a current response of 10 mA cm^{-2} .¹⁹² Oxygen doping to boost Atomic Hydrogen Binding Energy on Ni–Co–P for extremely efficient hydrogen synthesis was produced by Chunlei Liu and co-workers. They theoretically demonstrated how O-doping in Ni–Co–P may precisely modify the atomic hydrogen binding energy by DFT computations. To get an optimal atomic hydrogen binding energy, Ni–Co–P was strategically doped with oxygen. This is a powerful method that enhances intrinsic conductivity and augments exposed active areas to provide better HER activity. The exceptional HER activity results from the quick production of the overpotential (30 mV) of NiCo-phosphate at the surface after mild oxygen doping, plenty of exposed active regions and fine-tuning the atomic H₂ binding energy on cobalt and phosphorus sites. H–NiCoP NWAs/NF exhibited remarkable HER performance in neutral and alkaline media, with controlled O-doping outperforming most of the most advanced current binaries and tertiary phosphide catalysts. Thus, an O-doped Ni–Co–P (about 0.98% of O) nanowire array that has been tuned displayed a tiny Tafel slope of 38.6 mV dec^{-1} and an exceptionally low HER overpotential of 44 mV to drive 10 mA cm^{-2} .¹⁹³ To improve the evolving hydrogen methods, Ummul K. Sultana and co-workers developed O-doped cobalt sulfide. They found that the most successful electrochemical method was to repeatedly cycle the potential instead of holding it constant. This produced an amorphous CoS_x layer that was injected with oxygen. Based on calculations using DFT, it was shown that amorphous materials with a lower S–Co coordination number ideally increased the binding energy needed for hydrogen adsorption, which in turn promoted effective electron transfer kinetics. Additionally, under alkaline circumstances, this material showed activity for the HER, converting to cobalt oxide as shown, and producing a small Tafel slope of 67 mV dec^{-1} .¹⁹⁴ The O-doped HER electrocatalysts are summarised in Table 12.

5.5 S-doped electrocatalysts for HER

Sulfur atoms assist the adsorption and dissociation of hydrogen ions and electrons by adding more active sites to the catalyst surface. The HER kinetics are accelerated by this increase

in catalytic efficiency, leading to lower overpotential and higher responses with the sulfur. S-doping also makes the catalyst material more stable, reducing corrosion or structural deterioration throughout prolonged electrochemical operations. Through sulfidation reactions sulfur, a common chalcogen element, can be added as a dopant to the host catalyst's lattice structure. Sulfur dopant–electrolyte interaction is important because it influences charge movement kinetics at the catalyst–electrolyte interface, which helps to facilitate effective electron transport during the HER. Sulfur dopants can change the HER's rate of reaction, resulting in lower overpotential and quicker reaction rates. The catalyst's total energy effectiveness improves as a result of this kinetics enhancement, which facilitates more effective hydrogen production at lower energy inputs as well.^{198,207,208}

S-loaded RhNi alloys were discovered as very effective electrocatalysts for the hydrogen evolution process by Jiajia Lu and co-workers. They used a method involving hydrothermal heating and high-temperature annealing to successfully produce the worm-shaped S-doped RhNi material. The HER was improved in these S-doped RhNi compositions (S-RhNi) by combining doped heteroatoms, alloying with nonprecious metals, and taking on distinctive morphologies and architectures. With a starting potential at 17 mV at 20 mA cm^{-2} and a Tafel slope of $24.61 \text{ mV dec}^{-1}$, experimental results showed that S-RhNi shows Pt-like HER activity. The enhanced functionality was ascribed to the combined influence of the 3D arrangement of enriched S atoms and the worm-like RhNi material.¹⁹⁷ To generate the HER, Qijun Che and coworkers developed a unique approach called one-step electro-deposition of hierarchically organized S-doped NiCo films. This technique showed exceptional catalytic efficiency. Through the availability of many active sites, detected through XPS spectrum analysis, their inquiry into the ideal concentration of doped sulfur demonstrated its critical significance in lowering the overpotential of hydrogen evolution. Furthermore, electron transport was made easier by the development of metallic nickel and cobalt. The amorphous Co_xNi_yS material and the crystalline nickel and cobalt metals seemed to work in concert to increase HER function, shown in Fig. 17a. High-efficiency water electrolysis may be achieved using S-doping and careful component management. DFT study showed that S doping and component control are effective methods for high-efficiency water electro-

Table 12 O-doped electrocatalysts for HER

S. no	Catalyst	Electrolyte	Overpotential (η) at 10 mA cm^{-2}	Tafel slope (mV dec^{-1})	C_{dl} (mF cm^{-2})	Stability (hours & cycles)	Ref.
1	O-NiCoP/Ni ₂ P	1.0 M KOH	58	68.8	55.3	24 h	185
2	O-MoS ₂	0.5 M H ₂ SO ₄	120	55	37.7	3000 CV cycles	186
3	O-WS ₂ -1T	0.5 M H ₂ SO ₄	88	47	—	500 CV cycles	187
4	OMS/CC	0.5 M H ₂ SO ₄	300	58	—	2000 CV cycles	188
5	CC@O-CoFeS	0.5 M H ₂ SO ₄	105	62	9.13	24 h	189
6	6W–O–CoP	1.0 M KOH	84	61.1	95.6	10 000 CV cycles	190
7	VS ₄	0.5 M H ₂ SO ₄	48	44	2.82	75 h	191
8	O-Co ₂ P/CuO NWs/CF	1.0 M KOH	101	69.4	39.6	30 h	192
9	H-NiCoP NWAs/NF	1.0 M KOH	44	38.6	71	30 h	193
10	O-CoS _x	1.0 M KOH	—	72	—	24 h	194



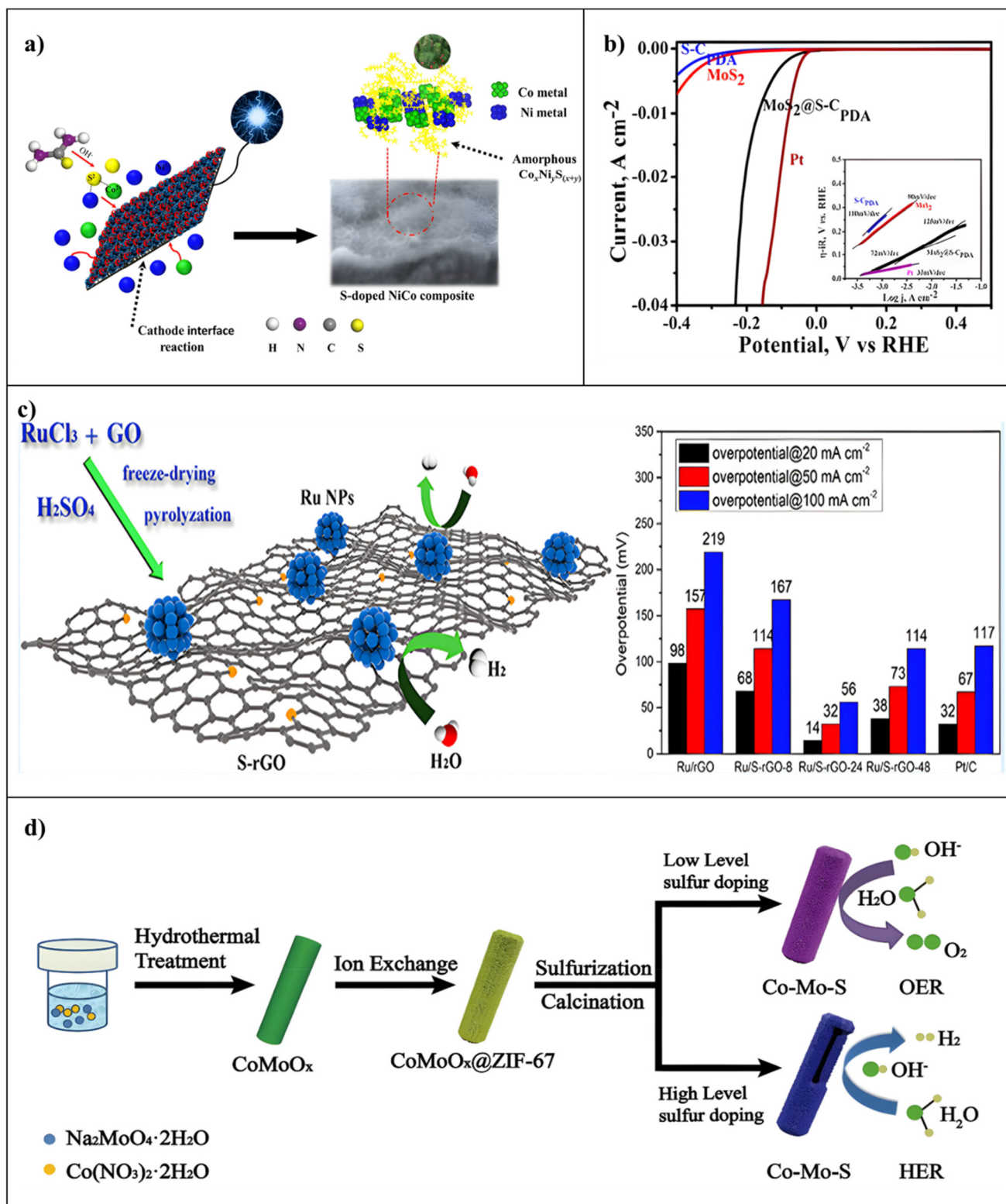


Fig. 17 (a) The electron-induced mechanism for the synthesis of S-doped NiCo composite. Reproduced from ref. 198 with permission. Copyright 2020, RSC. (b) MoS₂@S-CPDA, MoS₂, and SCPDA-modified electrodes in 0.5 M H₂SO₄ linear sweep voltammograms. Reproduced from ref. 205 with permission. Copyright 2015, RSC. (c) Ru/S-rGO synthesis process. Reproduced from ref. 201 with permission. Copyright 2020, ACS. (d) Illustration showing the steps involved in creating S-doped cobalt molybdate samples from Co–Mo precursor. Reproduced from ref. 203 with permission. Copyright 2018, ACS.



chemical reduction. With a hierarchical shape, the S-NiCo@50 electrode was notable for its very low overpotential of 28 mV at 10 mA cm⁻² and 125 mV at 100 mA cm⁻².¹⁹⁸ A binder-free electrocatalyst consisting of S-added Ni-P nanospheres was developed by Muhammad Aqeel Ashraf and co-workers using a new pulse electrochemical deposition method to improve total water splitting. Through a highly electrochemically active surface, powerful synergistic effects between the sulfur and phosphorus, fast gas separation from the surface, decreased resistance by bubble pinning, and enhanced wettability due to using nanostructuring, the resultant S-doped Ni-P small spheres demonstrated remarkable electrocatalytic capabilities for the HER. Among transition metal phosphides, Ni-Ps are the most notable because of their electrical conductivity, strength, and advantageous electrocatalytic capabilities. Ni-P phosphide-based samples are particularly significant because of their outstanding activity and stability. XPS analysis was used to analyze surface chemistry and element valence. DFT calculations show that doping resulted in a drop in ΔG_{H^+} , enhancing intrinsic electrocatalytic activity. With just a 55 mV overpotential needed to get a current density of 10 mA cm⁻² for the evolution of hydrogen reaction, the electrode astoundingly showed exceptional performance.¹⁹⁹ Xiaojiao Fang and co-workers synthesized S-loaded Co-Se₂ rich with the 1T phase, which they used to generate an effective electrocatalyst for the reaction of hydrogen evolution. Co-Se₂'s intrinsic metallicity makes it a promising electrocatalyst for breaking down water, but its chemically inert base plane limits its use by creating a shortage of active sites. By adjusting the sulfur doping level, they looked at how growth conditions affect hydrogen evolution and found that CoS_{0.1}Se_{1.9} with an enriched 1T-phase showed strong HER action. It took only 157 mV (vs. RHE) to reach a current density of 10 mA cm⁻² with a limited Tafel slope of 28.2 mV dec⁻¹.²⁰⁰ For the HER, Xuzhuo Sun and co-workers reported ruthenium nanoparticles widely distributed on sulfur-doped graphene. The results of their experiments demonstrated that the synthesis and improved distribution of ultrasmall Ru nanoparticles were enhanced by both sulfur treatment and a modest degree of oxidation of graphene oxide. This resulted in greater electrochemically active surface regions and increased exposure of active areas. XPS and electronic charge-difference investigations demonstrated that the metal-support relationship between S-doped graphene and Ru nanostructures led to a decrease in Ru's electron density, which in turn promoted electron release from H₂O and facilitated H-OH bond breakdown. Sulfur substances might accelerate water dissociation during the alkaline HER by lowering the energy barrier for breaking the H-OH bond, as shown by simulations using density functional theory. Interestingly, the catalyst had the lowest overpotential of 14 mV at a current density of 20 mA cm⁻², shown in Fig. 17c.²⁰¹ Ye Tian and co-workers developed sulfur-incorporated, plasma-etched graphene as a unique means of enhancing the hydrogen-evolving process. They utilized a technique called plasma engraving to produce more topological flaws in the S-doped graphene surface. At a current density of 10 mA cm⁻² the S-doped graphene that resulted was plasma-etched and showed markedly enhanced HER operation, exhibit-

ing an overpotential of 178 mV. XPS was used to evaluate the impact of plasma etching on the degree as well as type of doping in SG. The enhanced performance was ascribed to the mutually beneficial interaction between sulfur doping and topological errors caused by plasma. Moreover, the investigation showed that HER activity could be maximized by adjusting the sulfur doping amount in conjunction with thiophene-S-rich molecules and a suitable number of topological imperfections. This effective pairing of heteroatom doping and plasma-assisted defect engineering presents a viable path toward the production of extremely effective metal-free graphene-based catalysts for the HER. Xue Cui and co-workers developed a new method to generate a tuneable water-splitting catalyst by gradually modifying the doping of sulfur (S) in cobalt molybdate (CoMoO₄) by employing thioacetamide (TAA) as the precursor for sulfur. Notably, the S-doped TAA demonstrated electrocatalytic performance comparable to state-of-the-art transition-metal (Co, Ni, and Fe)-based HER electrodes, with a high TAA/CoMoO₄ mass ratio of 1.6 and outstanding selectivity for the HER, shown in Fig. 17d. In particular, it exhibited outstanding HER results, with an onset potential of 87 mV and a Tafel slope of 63 mV dec⁻¹.²⁰³ Xiang Zhang and co-workers developed AuPbPt Alloy Nanowire Networks doped with sulfur that act as very efficient HER catalysts. These networks are formed by depositing Platinum onto S-doped Au-Pb NWNs, which are substrates that come from the directed self-assembly of Pb²⁺ ions and the reduction of 2 nm gold nanoclusters by NaBH₄. XPS and DFT studies showed that the resultant S-doped Au-Pb-Pt alloy NWNs with an ideal PTS concentration had a considerably lower shift in the d-band center of platinum. By strengthening the active Pt sites' resistance to oxidation, this modification improved the overall stability of Pt surfaces as well as the catalytic activity toward HER. With an overpotential of 12 mV at 10 mA cm⁻² in acidic conditions, the as-synthesized S-doped Au-Pb-Pt alloy NWNs exhibited outstanding electrocatalytic activity for HER. They also exhibited amazing durability.²⁰⁴ To improve the evolution of hydrogen reaction performance, Thangavel Naresh Kumar and co-workers developed the dimensional and electrical refinement of MoS₂ nanosheets using S-doped carbon. When Mo(vi)-polydopamine (PDA) was pyrolyzed in the presence of elemental sulfur, S-doped carbon was integrated between MoS₂ nano-sheets, resulting in an improvement of MoS₂ with extra edge planes (2H 1T), which increased the charge density and conductive properties. The greater concentration of unsaturated S at the edge regions as opposed to the basal planes was responsible for the enhanced activity observed, even though bulk MoS₂'s intrinsic conductive properties and crystallinity constraints need significant overpotentials for HER. Additionally, the metallic 1T polymorph of MoS₂ compounds and the existence of edge sites were related to the catalytic properties. Recent research has shown that the metallic 1T (octahedral) phase of semi-conducting 2H MoS₂ (trigonal prismatic) considerably improved HER catalytic efficiency. At a current density of 10 mA cm⁻², the phase-transformed MoS₂[@S-CPDA] exhibited better HER activity with an overpotential of 160 mV, shown in Fig. 17b.²⁰⁵ Highly porous CoSe₂ nanosheets treated with sulfur



Table 13 S-doped electrocatalysts for HER

S. no	Catalyst	Electrolyte	Overpotential (η) at 10 mA cm ⁻²	Tafel slope (mV dec ⁻¹)	C_{dl} (mF cm ⁻²)	Stability (hours & cycles)	Ref.
1	S-RhNi	0.5 M H ₂ SO ₄	17 at (20 mA cm ⁻²)	24.61	—	10 h	197
2	S-NiCo@50	1.0 M NaOH	28	82	—	24 h	198
3	S-doped Ni-P	1.0 M KOH	55	70	16.5	25 h	199
4	CSSe-0.10	0.5 M H ₂ SO ₄	157	28.2	2.79	3000 CV cycles	200
5	Ru/S-rGO-24	1.0 M KOH	14 at (20 mA cm ⁻²)	38	49.01	20 h	201
6	SG-P	0.5 M H ₂ SO ₄	178 at (100 mA cm ⁻²)	86	16.5	3000 CV cycles	202
7	Co-Mo-S	1.0 M KOH	—	118	—	8 h	203
8	S-AuPbPt-NWNS	0.5 M H ₂ SO ₄	12	60.67	—	5000 CV cycles	204
9	MoS ₂ @S-CPDA	0.5 M H ₂ SO ₄	150	63	—	1000 CV cycles	205
10	S-CoSe ₂	0.5 M H ₂ SO ₄	102	50	—	4000 CV cycles	206

were developed by Ning Xue and co-workers as a powerful electrocatalyst for the hydrogen evolution process. An effective catalyst like CoSe₂ was made more active toward HER by adding foreign atoms like S to its structure. The solvothermal and ion exchange methods they employed to add sulfur to CoSe₂ nanosheets (NSs) were substantiated by theoretical and experimental studies, proving the efficacy of the resulting S-added CoSe₂ (S-CoSe₂) nanoporous NSs as a common and highly efficient catalyst for HER. High-resolution XPS spectra of S-CoSe₂-5 were acquired in the Se 3p region to illustrate the chemical bonding state of CoSe₂ before and after S doping. Based on DFT computations, it was revealed that S doping improved the efficiency of HER on S-CoSe₂ by reducing the kinetic barrier energy of the rate-determining step and decreasing the absolute value of ΔG_H , which moved closer to 0. At a low overpotential of 88 mV, the improved S-CoSe₂ catalyst achieved a catalytic current density of 10 mA cm⁻² for HER, making it one of the most effective catalysts based on CoSe and CoS for HER.²⁰⁶ The S-doped HER electrocatalysts are summarised in Table 13.

6. Summary and outlook

Heteroatom doping has emerged as one of the most effective and versatile strategies for engineering electrocatalysts with enhanced performance toward the HER. Through the deliberate incorporation of both metallic and non-metallic dopants into host materials, researchers have successfully tuned electronic structures, introduced catalytic active sites, modulated hydrogen binding energies, and improved conductivity. The dual advantages of electronic modification and structural activation offered by dopants allow for a fine-tuning of intrinsic catalytic activity and reaction kinetics, even under demanding electrochemical conditions. Metal dopants such as Fe, Co, Ni, Mo, Mn, Pt and Cu contribute primarily by shifting the d-band center and enhancing conductivity, while non-metal dopants like N, P, S, and B introduce surface defects, facilitate charge redistribution, and influence hydrogen adsorption energetics. The increasing interest in co-doping strategies arises from their synergistic effects, where simultaneous metallic and non-metallic doping generates multi-functional active sites, thereby overcoming the limitations posed by single-element doping.

These advances collectively contribute to higher catalytic efficiency, improved reaction dynamics, and enhanced durability of HER catalysts. Despite these achievements, challenges remain. The rational design of dopant combinations, atomic configurations, and optimal concentrations is still largely empirical. There is no universal framework to predict dopant performance due to the variability introduced by host substrates, synthesis conditions, and operating environments. Stability under industrially relevant conditions, particularly at high current densities and in harsh pH environments, remains a key hurdle. Moreover, bubble accumulation, dopant leaching, and structural degradation over extended periods continue to hinder practical deployment. Despite significant progress in doped electrocatalysts for the HER, certain challenges exist. Achieving precise control over dopant distribution and local coordination environments remains difficult, limiting reproducible and rational design; this can be addressed by new synthesis methodologies with improved atomic-level control. Furthermore, many doped catalysts undergo surface reconstruction or compositional changes when operated under HER conditions, requiring the use of *operando* and *in situ* characterisation approaches to determine the real active sites. Finally, establishing obvious structure-activity correlations remains difficult, necessitating further integration of experimental data with theoretical calculations in future research.

Looking forward, several research directions are poised to drive the next wave of innovation:

- **Design of model catalytic systems** that allow isolation of dopant effects and enable systematic investigations across various substrates.
- **Surface engineering strategies** to control wettability, reduce bubble adhesion and enhance gas release during HER operation.
- **Advanced synthesis techniques** such as atomic layer deposition and molecular precursor-based doping for controlled dopant placement and uniform distribution.
- **Application of external stimulation** including electric and magnetic fields to boost mass transport, reduce overpotentials and facilitate gas detachment.
- **Integration of computational and experimental insights**, particularly through machine learning, high-throughput DFT



calculations, and *in situ* spectroscopies to accelerate rational catalyst discovery.

• **Scalability and sustainability considerations** ensuring that synthesis routes are compatible with industrial needs and environmentally benign.

In summary, while heteroatom doping has already unlocked significant advancements in HER catalysis, future efforts should focus on combining atomic-level understanding with scalable technologies to bridge the gap between laboratory discoveries and commercial realization of green hydrogen production.

Author contributions

Manova Santhosh Yesupatham and Rajini Murugesan contributed equally to this paper. Manova Santhosh Yesupatham: conceptualization, writing – original draft (equal), methodology (equal), validation (equal). Rajini Murugesan: writing – original draft (equal), methodology (equal), validation (equal). Donald Richard: review & editing. Akshaya Radhakrishnan: review & editing. Maruthapillai Arthanareeswari: review & editing.

Conflicts of interest

There are no conflicts to declare.

Data availability

No primary research results, software or code have been included and no new data were generated or analyzed as part of this review.

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References

- 1 A. Kalair, N. Abas, M. S. Saleem, A. R. Kalair and N. Khan, *Energy Storage*, 2021, **3**, e135.
- 2 C. Gunathilake, I. Soliman, D. Panthi, P. Tandler, O. Fatani, N. A. Ghulamullah, D. Marasinghe, M. Farhath, T. Madhujith and K. Conrad, *Chem. Soc. Rev.*, 2024, **53**, 10900–10969.
- 3 X. Xu, Y. Pan, Y. Zhong, L. Ge, S. P. Jiang and Z. Shao, *Composites, Part B*, 2020, **198**, 108214.
- 4 M. S. Yesupatham, A. Augustin, N. Agamendran, B. Honnappa, M. Shanmugam, P. J. J. Sagayaraj, G. Thennarasu, N. C. S. Selvam and K. Sekar, *Sustainable Energy Fuels*, 2023, **7**, 4727–4757.
- 5 K. M. Y. Baig, S. Marimuthu, G. Maduraiveeran and G. K. Kole, *Dalton Trans.*, 2025, **54**, 11025–11035.
- 6 F. Abdelghafar, X. Xu and Z. Shao, *Mater. Rep.: Energy*, 2022, **2**, 100144.
- 7 A. Mehtab, S. A. Ali, I. Sadiq, S. Shaheen, H. Khan, M. Fazil, N. A. Pandit, F. Naaz and T. Ahmad, *ACS Sustainable Resour. Manage.*, 2024, **1**, 604–620.
- 8 A. K. Diaz-Duran, G. Iadarola-Perez, E. B. Halac and F. Roncaroli, *Top. Catal.*, 2022, **65**, 887.
- 9 R. Murugesan, N. C. S. Selvam and A. Maruthapillai, *Catal. Sci. Technol.*, 2025, **15**, 6928–6975.
- 10 H. Lu, J. Tournet, K. Dastafkan, Y. Liu, Y. H. Ng, S. K. Karuturi, C. Zhao and Z. Yin, *Chem. Rev.*, 2021, **121**, 10271–10366.
- 11 K. M. Y. Baig, S. Marimuthu, E. Zangrando, G. Maduraiveeran and G. K. Kole, *Mater. Today Chem.*, 2025, **43**, 102464.
- 12 J. Wang, Y. Gao, H. Kong, J. Kim, S. Choi, F. Ciucci, Y. Hao, S. Yang, Z. Shao and J. Lim, *Chem. Soc. Rev.*, 2020, **49**, 9154–9196.
- 13 Y. Liu, R. Duan, X. Li, L. Luo, J. Gong, G. Zhang, Y. Li and Z. Li, *Inorg. Chem.*, 2022, **61**, 13210–13217.
- 14 J. Wang, T. Liao, Z. Wei, J. Sun, J. Guo and Z. Sun, *Small Methods*, 2021, **5**, 2000988.
- 15 A. Majumdar, K. D. Tran, D. Malhotra, D. T. Tran, N. H. Kim and J. H. Lee, *Heteroatom-Doped Carbon Allotropes: Progress in Synthesis, Characterization, and Applications*, 2024, pp. 177–222.
- 16 A. Zhang, Y. Liang, H. Zhang, Z. Geng and J. Zeng, *Chem. Soc. Rev.*, 2021, **50**, 9817–9844.
- 17 R. Murugesan, M. S. Yesupatham, N. Agamendran, K. Sekar, C. S. S. Neethinathan and A. Maruthapillai, *Energy Technol.*, 2024, 2400882.
- 18 Y. Zhao, W. Jiang, J. Zhang, E. C. Lovell, R. Amal, Z. Han and X. Lu, *Adv. Mater.*, 2021, **33**, 2102801.
- 19 M. Du, F. Yu, S. Gong and F. Liu, *Adv. Funct. Mater.*, 2025, **35**, 2413826.
- 20 T. Guo, Y. Lin, X. Chen, J. Lu, X. Zhao, X. Yao, L. Meng, Y. Liu and X. Zhang, *Int. J. Hydrogen Energy*, 2023, **48**, 2990–2997.
- 21 Q. Cai, W. Hong, C. Jian, X. He and W. Liu, *Adv. Energy Sustainability Res.*, 2023, **4**, 2200178.
- 22 W. Chen, M.-K. Zhang, B.-Y. Liu, J. Cai and Y.-X. Chen, *Curr. Opin. Electrochem.*, 2022, **34**, 101003.
- 23 Y. Li, X. Wei, L. Chen and J. Shi, *Angew. Chem., Int. Ed.*, 2021, **60**, 19550–19571.
- 24 A. Kahyarian, B. Brown and S. Nestic, *J. Electrochem. Soc.*, 2017, **164**, H365.
- 25 N. Mahmood, Y. Yao, J. Zhang, L. Pan, X. Zhang and J. Zou, *Adv. Sci.*, 2018, **5**, 1700464.



- 26 F. Li, G.-F. Han, H.-J. Noh, J.-P. Jeon, I. Ahmad, S. Chen, C. Yang, Y. Bu, Z. Fu and Y. Lu, *Nat. Commun.*, 2019, **10**, 4060.
- 27 P. F. Liu, H. Yin, H. Q. Fu, M. Y. Zu, H. G. Yang and H. Zhao, *J. Mater. Chem. A*, 2020, **8**, 10096–10129.
- 28 M. Boshir Ahmed, J. Alom, M. S. Hasan, M. Asaduzzaman, M. S. Rahman, R. Hossen, M. Abu Hasan Johir, M. Taufiq Alam, J. L. Zhou and Y. Zhu, *ChemNanoMat*, 2023, **9**, e202200482.
- 29 Y. Pan, T. Jian, P. Gu, Y. Song, Q. Wang, B. Han, Y. Ran, Z. Pan, Y. Li and W. Xu, *Nat. Commun.*, 2024, **15**, 9631.
- 30 Y. Qin, G. Lu, F. Yang, C. Xu, S. Jiang, Y. Wang, Y. Tang and P. Wang, *Mater. Adv.*, 2023, **4**, 1226–1248.
- 31 S. Gbadamasi, M. Mohiuddin, V. Krishnamurthi, R. Verma, M. W. Khan, S. Pathak, K. Kalantar-Zadeh and N. Mahmood, *Chem. Soc. Rev.*, 2021, **50**, 4684–4729.
- 32 R. Paul, F. Du, L. Dai, Y. Ding, Z. L. Wang, F. Wei and A. Roy, *Adv. Mater.*, 2019, **31**, 1805598.
- 33 S. Kumar, R. Sharma, S. J. Borah, A. Gupta, M. K. Gupta, R. Kumar, K. K. Dubey, Y. K. Mishra and V. Kumar, *Sustainable Energy Fuels*, 2023, **7**, 4354–4395.
- 34 X. Feng, Y. Bai, M. Liu, Y. Li, H. Yang, X. Wang and C. Wu, *Energy Environ. Sci.*, 2021, **14**, 2036–2089.
- 35 A. Zunger and O. I. Malyi, *Chem. Rev.*, 2021, **121**, 3031–3060.
- 36 A. Ghosh, O. F. Mohammed and O. M. Bakr, *Acc. Chem. Res.*, 2018, **51**, 3094–3103.
- 37 N. R. Abdullah, M. T. Kareem, H. O. Rashid, A. Manolescu and V. Gudmundsson, *Phys. E*, 2021, **129**, 114644.
- 38 R. Kronberg, H. Lappalainen and K. Laasonen, *J. Phys. Chem. C*, 2021, **125**, 15918–15933.
- 39 C. Wei, S. Sun, D. Mandler, X. Wang, S. Z. Qiao and Z. J. Xu, *Chem. Soc. Rev.*, 2019, **48**, 2518–2534.
- 40 J.-F. Chen, Y. Mao, H.-F. Wang and P. Hu, *ACS Catal.*, 2016, **6**, 6804–6813.
- 41 X. Du, J. Huang, J. Zhang, Y. Yan, C. Wu, Y. Hu, C. Yan, T. Lei, W. Chen and C. Fan, *Angew. Chem., Int. Ed.*, 2019, **58**, 4484–4502.
- 42 Z. Chen, H. Yang, Z. Kang, M. Driess and P. W. Menezes, *Adv. Mater.*, 2022, **34**, 2108432.
- 43 G. Tian, Q. Zhang, B. Zhang, Y. Jin, J. Huang, D. S. Su and F. Wei, *Adv. Funct. Mater.*, 2014, **24**, 5956–5961.
- 44 Y. Wang, X. Duan, Y. Xie, H. Sun and S. Wang, *ACS Catal.*, 2020, **10**, 13383–13414.
- 45 J. Wu, Q. Zhang, K. Shen, R. Zhao, W. Zhong, C. Yang, H. Xiang, X. Li and N. Yang, *Adv. Funct. Mater.*, 2022, **32**, 2107802.
- 46 Y. Ma, M. Chen, H. Geng, H. Dong, P. Wu, X. Li, G. Guan and T. Wang, *Adv. Funct. Mater.*, 2020, **30**, 2000561.
- 47 J. Xie, J. Qi, F. Lei and Y. Xie, *Chem. Commun.*, 2020, **56**, 11910–11930.
- 48 W.-J. Jiang, T. Tang, Y. Zhang and J.-S. Hu, *Acc. Chem. Res.*, 2020, **53**, 1111–1123.
- 49 M. A. Bhat and K. Majid, *Energy Fuels*, 2023, **37**, 18834–18842.
- 50 H. Bai, Y. Han, X. Rong, Y. Yu, J. Ma, T. Yang, H. Huang and J. Hu, *Int. J. Hydrogen Energy*, 2023, **48**, 20562–20576.
- 51 H. Jin, X. Liu, S. Chen, A. Vasileff, L. Li, Y. Jiao, L. Song, Y. Zheng and S.-Z. Qiao, *ACS Energy Lett.*, 2019, **4**, 805–810.
- 52 B. Bayatsarmadi, Y. Zheng, A. Vasileff and S. Qiao, *Small*, 2017, **13**, 1700191.
- 53 Z. Li, X. Lu, J. Teng, Y. Zhou and W. Zhuang, *Nanoscale*, 2021, **13**, 11314–11324.
- 54 J. Shan, C. Ye, C. Zhu, J. Dong, W. Xu, L. Chen, Y. Jiao, Y. Jiang, L. Song and Y. Zhang, *J. Am. Chem. Soc.*, 2022, **144**, 23214–23222.
- 55 M. Zhou, C. Li and J. Fang, *Chem. Rev.*, 2020, **121**, 736–795.
- 56 J. Li, C. A. Triana, W. Wan, D. P. A. Saseendran, Y. Zhao, S. E. Balaghi, S. Heidari and G. R. Patzke, *Chem. Soc. Rev.*, 2021, **50**, 2444–2485.
- 57 F. Liu, C. Shi, X. Guo, Z. He, L. Pan, Z. Huang, X. Zhang and J. Zou, *Adv. Sci.*, 2022, **9**, 2200307.
- 58 T. Shi, B. Gao, H. Meng, Y. Fu, D. Kong, P. Ren, H. Fu and Z. Feng, *Green Chem.*, 2024, **26**, 4209–4220.
- 59 Z. Li, X. Li, Y. Gu, X. Hu, L. Wang and P. Li, *Appl. Surf. Sci.*, 2024, **646**, 158959.
- 60 X. Liu, Q. Yu, X. Qu, X. Wang, J. Chi and L. Wang, *Adv. Mater.*, 2024, **36**, 2307395.
- 61 X. Guo, X. Yu, Z. Feng, J. Liang, Q. Li, Z. Lv, B. Liu, C. Hao and G. Li, *ACS Sustainable Chem. Eng.*, 2018, **6**, 8150–8158.
- 62 M. Guo, S. Song, S. Zhang, Y. Yan, K. Zhan, J. Yang and B. Zhao, *ACS Sustainable Chem. Eng.*, 2020, **8**, 7436–7444.
- 63 C. Tang, R. Zhang, W. Lu, L. He, X. Jiang, A. M. Asiri and X. Sun, *Adv. Mater.*, 2017, **29**, 1602441.
- 64 T. H. Wondimu, G.-C. Chen, D. M. Kabtamu, H.-Y. Chen, A. W. Bayeh, H.-C. Huang and C. H. Wang, *Int. J. Hydrogen Energy*, 2018, **43**, 6481–6490.
- 65 J. Yan, H. Wu, H. Chen, R. Jiang and S. F. Liu, *J. Mater. Chem. A*, 2017, **5**, 10173–10181.
- 66 C. Wan and B. M. Leonard, *Chem. Mater.*, 2015, **27**, 4281–4288.
- 67 F. Li, Y. Bu, Z. Lv, J. Mahmood, G. Han, I. Ahmad, G. Kim, Q. Zhong and J. Baek, *Small*, 2017, **13**, 1701167.
- 68 P. Wang, Z. Pu, Y. Li, L. Wu, Z. Tu, M. Jiang, Z. Kou, I. S. Amiinu and S. Mu, *ACS Appl. Mater. Interfaces*, 2017, **9**, 26001–26007.
- 69 C. Lin, P. Wang, H. Jin, J. Zhao, D. Chen, S. Liu, C. Zhang and S. Mu, *Dalton Trans.*, 2019, **48**, 16555–16561.
- 70 H. Fan, H. Yu, Y. Zhang, Y. Zheng, Y. Luo, Z. Dai, B. Li, Y. Zong and Q. Yan, *Angew. Chem., Int. Ed.*, 2017, **56**, 12566–12570.
- 71 Y. Liu, Y. Yang, B. Chen, X. Li, M. Guo, Y. Yang, K. Xu and C. Yuan, *Inorg. Chem.*, 2021, **60**, 18325–18336.
- 72 H. Xie, C. Lan, B. Chen, F. Wang and T. Liu, *Nano Res.*, 2020, **13**, 3321–3329.
- 73 T. Liu, P. Li, N. Yao, G. Cheng, S. Chen, W. Luo and Y. Yin, *Angew. Chem.*, 2019, **131**, 4727–4732.
- 74 D.-Y. Wang, M. Gong, H.-L. Chou, C.-J. Pan, H.-A. Chen, Y. Wu, M.-C. Lin, M. Guan, J. Yang and C.-W. Chen, *J. Am. Chem. Soc.*, 2015, **137**, 1587–1592.



- 75 G. Zhang, X. Zheng, Q. Xu, J. Zhang, W. Liu and J. Chen, *J. Mater. Chem. A*, 2018, **6**, 4793–4800.
- 76 B. Seo, G. Y. Jung, Y. J. Sa, H. Y. Jeong, J. Y. Cheon, J. H. Lee, H. Y. Kim, J. C. Kim, H. S. Shin and S. K. Kwak, *ACS Nano*, 2015, **9**, 3728–3739.
- 77 H. Lin, N. Liu, Z. Shi, Y. Guo, Y. Tang and Q. Gao, *Adv. Funct. Mater.*, 2016, **26**, 5590–5598.
- 78 I. Roger, R. Moca, H. N. Miras, K. G. Crawford, D. A. J. Moran, A. Y. Ganin and M. D. Symes, *J. Mater. Chem. A*, 2017, **5**, 1472–1480.
- 79 F. Wang, Y. Zhu, W. Tian, X. Lv, H. Zhang, Z. Hu, Y. Zhang, J. Ji and W. Jiang, *J. Mater. Chem. A*, 2018, **6**, 10490–10496.
- 80 X. Dai, K. Du, Z. Li, M. Liu, Y. Ma, H. Sun, X. Zhang and Y. Yang, *ACS Appl. Mater. Interfaces*, 2015, **7**, 27242–27253.
- 81 W. Zhao, S. Wang, C. Feng, H. Wu, L. Zhang and J. Zhang, *ACS Appl. Mater. Interfaces*, 2018, **10**, 40491–40499.
- 82 P. P. Patel, O. I. Velikokhatnyi, S. D. Ghadge, P. J. Hanumantha, M. K. Datta, R. Kuruba, B. Gattu, P. M. Shanthi and P. N. Kumta, *Int. J. Hydrogen Energy*, 2018, **43**, 7855–7871.
- 83 Y. Zheng, J. Rong, J. Xu, Y. Zhu, T. Zhang, D. Yang and F. Qiu, *Appl. Surf. Sci.*, 2021, **563**, 150385.
- 84 X.-Y. Yu, Y. Feng, Y. Jeon, B. Guan, X. W. Lou and U. Paik, *Adv. Mater.*, 2016, **28**, 9006–9011.
- 85 X. Peng, A. M. Qasim, W. Jin, L. Wang, L. Hu, Y. Miao, W. Li, Y. Li, Z. Liu and K. Huo, *Nano Energy*, 2018, **53**, 66–73.
- 86 K. Xiong, L. Li, L. Zhang, W. Ding, L. Peng, Y. Wang, S. Chen, S. Tan and Z. Wei, *J. Mater. Chem. A*, 2015, **3**, 1863–1867.
- 87 Z. Liu, H. Tan, J. Xin, J. Duan, X. Su, P. Hao, J. Xie, J. Zhan, J. Zhang and J.-J. Wang, *ACS Appl. Mater. Interfaces*, 2018, **10**, 3699–3706.
- 88 W. Fang, D. Liu, Q. Lu, X. Sun and A. M. Asiri, *Electrochem. Commun.*, 2016, **63**, 60–64.
- 89 J. Ding, S. Ji, H. Wang, H. Gai, F. Liu, V. Linkov and R. Wang, *Int. J. Hydrogen Energy*, 2019, **44**, 2832–2840.
- 90 M. I. Abdullah, A. Hameed, N. Zhang and M. Ma, *Int. J. Hydrogen Energy*, 2019, **44**, 14869–14876.
- 91 Y. Yang, X. Zhao, H. Mao, R. Ning, X. Zheng, J. Sui and W. Cai, *Int. J. Hydrogen Energy*, 2020, **45**, 10724–10728.
- 92 T. Chen, B. Ye, H. Dai, S. Qin, Y. Zhang and Q. Yang, *J. Solid State Chem.*, 2021, **301**, 122299.
- 93 X. F. Lu, L. Yu and X. W. Lou, *Sci. Adv.*, 2019, **5**, eaav6009.
- 94 A. Kumar, S. Kumar, S. Jana and R. Prakash, *Energy Fuels*, 2023, **37**, 4552–4565.
- 95 Q. Luo, C. Wang, H. Xin, Y. Qi, Y. Zhao, J. Sun and F. Ma, *ACS Sustainable Chem. Eng.*, 2021, **9**, 732–742.
- 96 C. Wu, B. Liu, J. Wang, Y. Su, H. Yan, C. Ng, C. Li and J. Wei, *Appl. Surf. Sci.*, 2018, **441**, 1024–1033.
- 97 Y. Sun, L. Hang, Q. Shen, T. Zhang, H. Li, X. Zhang, X. Lyu and Y. Li, *Nanoscale*, 2017, **9**, 16674–16679.
- 98 Z. Wang, X. Ge, Z. Li, J. Wu, Z. Liang and S. Wang, *New J. Chem.*, 2019, **43**, 9652–9657.
- 99 Y. Zhou, J. Zhang, H. Ren, Y. Pan, Y. Yan, F. Sun, X. Wang, S. Wang and J. Zhang, *Appl. Catal., B*, 2020, **268**, 118467.
- 100 Z. Cui, Y. Ge, H. Chu, R. Baines, P. Dong, J. Tang, Y. Yang, P. M. Ajayan, M. Ye and J. Shen, *J. Mater. Chem. A*, 2017, **5**, 1595–1602.
- 101 L. Li, Y. Guo, X. Wang, X. Liu and Y. Lu, *Langmuir*, 2021, **37**, 5986–5992.
- 102 Q. Wang, H. Zhao, F. Li, W. She, X. Wang, L. Xu and H. Jiao, *J. Mater. Chem. A*, 2019, **7**, 7636–7643.
- 103 C. Du, Y. Men, X. Hei, J. Yu, G. Cheng and W. Luo, *ChemElectroChem*, 2018, **5**, 2564–2570.
- 104 X. Lu, M. Cai, J. Huang and C. Xu, *J. Colloid Interface Sci.*, 2020, **562**, 307–312.
- 105 X. Zhong, Y. Sun, X. Chen, G. Zhuang, X. Li and J. Wang, *Adv. Funct. Mater.*, 2016, **26**, 5778–5786.
- 106 I. S. Amini, Z. Pu, X. Liu, K. A. Owusu, H. G. R. Monestel, F. O. Boakye, H. Zhang and S. Mu, *Adv. Funct. Mater.*, 2017, **27**, 1702300.
- 107 T. Sun, J. Wang, X. Chi, Y. Lin, Z. Chen, X. Ling, C. Qiu, Y. Xu, L. Song and W. Chen, *ACS Catal.*, 2018, **8**, 7585–7592.
- 108 T. Liu, X. Ma, D. Liu, S. Hao, G. Du, Y. Ma, A. M. Asiri, X. Sun and L. Chen, *ACS Catal.*, 2017, **7**, 98–102.
- 109 M. Wang, Y. Tuo, X. Li, Q. Hua, F. Du and L. Jiang, *ACS Sustainable Chem. Eng.*, 2019, **7**, 12419–12427.
- 110 X. Wang, H. Zhou, D. Zhang, M. Pi, J. Feng and S. Chen, *J. Power Sources*, 2018, **387**, 1–8.
- 111 Y. Liu, X. Hua, C. Xiao, T. Zhou, P. Huang, Z. Guo, B. Pan and Y. Xie, *J. Am. Chem. Soc.*, 2016, **138**, 5087–5092.
- 112 F. Yu, Y. Gao, Z. Lang, Y. Ma, L. Yin, J. Du, H. Tan, Y. Wang and Y. Li, *Nanoscale*, 2018, **10**, 6080–6087.
- 113 Y. Zhang, Y. Liu, M. Ma, X. Ren, Z. Liu, G. Du, A. M. Asiri and X. Sun, *Chem. Commun.*, 2017, **53**, 11048–11051.
- 114 X. Li, S. Li, A. Yoshida, S. Sirisomboonchai, K. Tang, Z. Zuo, X. Hao, A. Abudula and G. Guan, *Catal. Sci. Technol.*, 2018, **8**, 4407–4412.
- 115 L. Zhang, M. Li, A. Zou, S. H. Yu, T. Xiong, L. Wang, J. He, Q. Fu, K. Sun and D. H. C. Chua, *ACS Appl. Energy Mater.*, 2018, **2**, 493–502.
- 116 S. Xu, X. Yu, X. Liu, C. Teng, Y. Du and Q. Wu, *J. Colloid Interface Sci.*, 2020, **577**, 379–387.
- 117 L. Wu, X. Xu, Y. Zhao, K. Zhang, Y. Sun, T. Wang, Y. Wang, W. Zhong and Y. Du, *Appl. Surf. Sci.*, 2017, **425**, 470–477.
- 118 Q. Sun, Y. Dong, Z. Wang, S. Yin and C. Zhao, *Small*, 2018, **14**, 1704137.
- 119 J. Dai, D. Zhao, W. Sun, X. Zhu, L.-J. Ma, Z. Wu, C. Yang, Z. Cui, L. Li and S. Chen, *ACS Catal.*, 2019, **9**, 10761–10772.
- 120 J. Zhang, B. Xiao, X. Liu, P. Liu, P. Xi, W. Xiao, J. Ding, D. Gao and D. Xue, *J. Mater. Chem. A*, 2017, **5**, 17601–17608.
- 121 Y. Ding, X. Du and X. Zhang, *ChemCatChem*, 2021, **13**, 1824–1833.
- 122 L. Wen, Y. Sun, C. Zhang, J. Yu, X. Li, X. Lyu, W. Cai and Y. Li, *ACS Appl. Energy Mater.*, 2018, **1**, 3835–3842.



- 123 L. Ji, P. Yan, C. Zhu, C. Ma, W. Wu, C. Wei, Y. Shen, S. Chu, J. Wang and Y. Du, *Appl. Catal., B*, 2019, **251**, 87–93.
- 124 H. Su, S. Song, S. Li, Y. Gao, L. Ge, W. Song, T. Ma and J. Liu, *Appl. Catal., B*, 2021, **293**, 120225.
- 125 Y. Li, X. Tan, R. K. Hocking, X. Bo, H. Ren, B. Johannessen, S. C. Smith and C. Zhao, *Nat. Commun.*, 2020, **11**, 2720.
- 126 Q. Sun, L. Wang, Y. Shen, M. Zhou, Y. Ma, Z. Wang and C. Zhao, *ACS Sustainable Chem. Eng.*, 2018, **6**, 12746–12754.
- 127 H. Li, G. Gao, H. Zhao, W. Wang, Y. Yang, Y. Du, S. Li, Y. Liu and L. Wang, *Int. J. Hydrogen Energy*, 2021, **46**, 33078–33086.
- 128 Y. Tian, L. Cao and P. Qin, *ChemCatChem*, 2019, **11**, 4420–4426.
- 129 L. Yan, B. Zhang, J. Zhu, Y. Li, P. Tsiakaras and P. K. Shen, *Appl. Catal., B*, 2020, **265**, 118555.
- 130 P. Bhanja, B. Mohanty, A. K. Patra, S. Ghosh, B. K. Jena and A. Bhaumik, *ChemCatChem*, 2019, **11**, 583–592.
- 131 Y. Li, Q. Gu, B. Johannessen, Z. Zheng, C. Li, Y. Luo, Z. Zhang, Q. Zhang, H. Fan and W. Luo, *Nano Energy*, 2021, **84**, 105898.
- 132 X. Xiao, D. Sun, X. Liu, B. Qiu, X. Xu, D. Zhang and T. Yang, *Sustainable Energy Fuels*, 2021, **5**, 1059–1066.
- 133 S. Anantharaj, P. E. Karthik, B. Subramanian and S. Kundu, *ACS Catal.*, 2016, **6**, 4660–4672.
- 134 W. Chen, X. Zhu, W. Wei, H. Chen, T. Dong, R. Wang, M. Liu, K. Ostrikov, P. Peng and S. Zang, *Small*, 2023, **19**, 2304294.
- 135 H. Hu, Y. Zheng, Y. Zhu, J. Rong, Y. Dai, T. Zhang, D. Yang and F. Qiu, *Inorg. Chem.*, 2022, **62**, 601–608.
- 136 Y. Sun, Y. Zang, W. Tian, X. Yu, J. Qi, L. Chen, X. Liu and H. Qiu, *Energy Environ. Sci.*, 2022, **15**, 1201–1210.
- 137 A. Abdolmaleki, Z. Mohamadi, A. A. Ensafi, N. Z. Atashbar and B. Rezaei, *Int. J. Hydrogen Energy*, 2018, **43**, 8323–8332.
- 138 A. Fan, C. Qin, X. Zhang, J. Yang, J. Ge, S. Wang, X. Yuan, S. Wang and X. Dai, *J. Mater. Chem. A*, 2019, **7**, 24347–24355.
- 139 M. K. Kundu, T. Bhowmik, R. Mishra and S. Barman, *ChemSusChem*, 2018, **11**, 2388–2401.
- 140 J. Li, J. Zhang, J. Zhang, K. Pan, H. Xu, H. Chen, G. Liu, N. Wu, C. Yuan and X. Liu, *J. Mater. Chem. A*, 2023, **11**, 19812–19844.
- 141 S. A. Singh, K. Vishwanath and G. Madras, *ACS Appl. Mater. Interfaces*, 2017, **9**, 19380–19388.
- 142 X. Cheng, Y. Li, L. Zheng, Y. Yan, Y. Zhang, G. Chen, S. Sun and J. Zhang, *Energy Environ. Sci.*, 2017, **10**, 2450–2458.
- 143 Y. P. Zhu, C. Guo, Y. Zheng and S.-Z. Qiao, *Acc. Chem. Res.*, 2017, **50**, 915–923.
- 144 Q. Hu, G. Li, Z. Han, Z. Wang, X. Huang, H. Yang, Q. Zhang, J. Liu and C. He, *Chem. – Eur. J.*, 2020, **26**, 3930–3942.
- 145 S. P. Kaur and T. J. D. Kumar, *Appl. Surf. Sci.*, 2021, **552**, 149146.
- 146 H. Hu, Z. Shi, K. Khan, R. Cao, W. Liang, A. K. Tareen, Y. Zhang, W. Huang, Z. Guo and X. Luo, *J. Mater. Chem. A*, 2020, **8**, 5421–5441.
- 147 C. Peng, L. Song, L. Wang, F. Yang, J. Ding, F. Huang and Y. Wang, *ACS Appl. Energy Mater.*, 2021, **4**, 4887–4896.
- 148 Y. Yoon, A. P. Tiwari, M. Choi, T. G. Novak, W. Song, H. Chang, T. Zyung, S. S. Lee, S. Jeon and K. An, *Adv. Funct. Mater.*, 2019, **29**, 1903443.
- 149 W. Fu, Y. Wang, W. Tian, H. Zhang, J. Li, S. Wang and Y. Wang, *Angew. Chem.*, 2020, **132**, 23999–24007.
- 150 L. Bian, W. Gao, J. Sun, M. Han, F. Li, Z. Gao, L. Shu, N. Han, Z. Yang and A. Song, *ChemCatChem*, 2018, **10**, 1571–1577.
- 151 H. Gu, W. Fan and T. Liu, *Nanoscale Horiz.*, 2017, **2**, 277–283.
- 152 H. Yan, C. Tian, L. Wang, A. Wu, M. Meng, L. Zhao and H. Fu, *Angew. Chem.*, 2015, **127**, 6423–6427.
- 153 Z. Shi, K. Nie, Z.-J. Shao, B. Gao, H. Lin, H. Zhang, B. Liu, Y. Wang, Y. Zhang and X. Sun, *Energy Environ. Sci.*, 2017, **10**, 1262–1271.
- 154 J. Zhang, Y. Liu, B. Xia, C. Sun, Y. Liu, P. Liu and D. Gao, *Electrochim. Acta*, 2018, **259**, 955–961.
- 155 L. Li, T. Zhang, J. Yan, X. Cai and S. Liu, *Small*, 2017, **13**, 1700441.
- 156 S. Li, N. Yang, L. Liao, Y. Luo, S. Wang, F. Cao, W. Zhou, D. Huang and H. Chen, *ACS Appl. Mater. Interfaces*, 2018, **10**, 37038–37045.
- 157 J.-S. Li, S. Zhang, J.-Q. Sha, H. Wang, M.-Z. Liu, L.-X. Kong and G.-D. Liu, *ACS Appl. Mater. Interfaces*, 2018, **10**, 17140–17146.
- 158 Y. Hu, H. Yu, L. Qi, J. Dong, P. Yan, T. Taylor Isimjan and X. Yang, *ChemSusChem*, 2021, **14**, 1565–1573.
- 159 Z. Zhou, L. Zeng, G. Xiong, L. Yang, H. Yuan, J. Yu, S. Xu, D. Wang, X. Zhang and H. Liu, *Chem. Eng. J.*, 2021, **426**, 129214.
- 160 P. Chen, T. Zhou, M. Chen, Y. Tong, N. Zhang, X. Peng, W. Chu, X. Wu, C. Wu and Y. Xie, *ACS Catal.*, 2017, **7**, 7405–7411.
- 161 Y. Wu, X. Liu, D. Han, X. Song, L. Shi, Y. Song, S. Niu, Y. Xie, J. Cai and S. Wu, *Nat. Commun.*, 2018, **9**, 1425.
- 162 J. Yan, L. Li, Y. Ji, P. Li, L. Kong, X. Cai, Y. Li, T. Ma and S. F. Liu, *J. Mater. Chem. A*, 2018, **6**, 12532–12540.
- 163 H. Jiang, J. Gu, X. Zheng, M. Liu, X. Qiu, L. Wang, W. Li, Z. Chen, X. Ji and J. Li, *Energy Environ. Sci.*, 2019, **12**, 322–333.
- 164 C. C. Yang, S. F. Zai, Y. T. Zhou, L. Du and Q. Jiang, *Adv. Funct. Mater.*, 2019, **29**, 1901949.
- 165 Y. Men, P. Li, F. Yang, G. Cheng, S. Chen and W. Luo, *Appl. Catal., B*, 2019, **253**, 21–27.
- 166 Y. Pan, Y. Liu, Y. Lin and C. Liu, *ACS Appl. Mater. Interfaces*, 2016, **8**, 13890–13901.
- 167 C. Pi, C. Huang, Y. Yang, H. Song, X. Zhang, Y. Zheng, B. Gao, J. Fu, P. K. Chu and K. Huo, *Appl. Catal., B*, 2020, **263**, 118358.
- 168 S. A. Shah, X. Shen, M. Xie, G. Zhu, Z. Ji, H. Zhou, K. Xu, X. Yue, A. Yuan and J. Zhu, *Small*, 2019, **15**, 1804545.



- 169 J. Ma, M. Wang, G. Lei, G. Zhang, F. Zhang, W. Peng, X. Fan and Y. Li, *Small*, 2018, **14**, 1702895.
- 170 C. Das, N. Sinha and P. Roy, *Small*, 2022, **18**, 2202033.
- 171 M. Li, L. Zhang, Q. Xu, J. Niu and Z. Xia, *J. Catal.*, 2014, **314**, 66–72.
- 172 S. Keshipour and F. Eyvari-Ashnak, *ChemElectroChem*, 2023, **10**, e202201153.
- 173 T. Li, J. Liu, Y. Song and F. Wang, *ACS Catal.*, 2018, **8**, 8450–8458.
- 174 B. R. Sathe, X. Zou and T. Asefa, *Catal. Sci. Technol.*, 2014, **4**, 2023–2030.
- 175 Z. Liu, D. Gao, L. Hu, H. Liu, Y. Li, Y. Xue, F. Liu, J. Zhang and C. Tang, *Colloids Surf., A*, 2022, **646**, 128903.
- 176 B. Adegbemiga Yusuf, M. Xie, W. Yaseen, C. Judith Oluigbo, W. Wei, Y. Xu and J. Xie, *ChemElectroChem*, 2021, **8**, 1337–1348.
- 177 Q. Lin, C. Shang, Z. Chen, X. Wang and G. Zhou, *Int. J. Hydrogen Energy*, 2020, **45**, 30659–30665.
- 178 M. Bat-Erdene, M. Batmunkh, B. Sainbileg, M. Hayashi, A. S. R. Bati, J. Qin, H. Zhao, Y. L. Zhong and J. G. Shapter, *Small*, 2021, **17**, 2102218.
- 179 Y. B. Adegbemiga, N. Ullah, M. Xie, S. Hussain, C. J. Oluigbo, W. Yaseen, A. J. Kumar, Y. Xu and J. Xie, *J. Alloys Compd.*, 2020, **835**, 155267.
- 180 H. Yang, Z. Chen, P. Guo, B. Fei and R. Wu, *Appl. Catal., B*, 2020, **261**, 118240.
- 181 Y. Gao, S. Qian, H. Wang, W. Yuan, Y. Fan, N. Cheng, H. Xue, T. Jiang and J. Tian, *Appl. Catal., B*, 2023, **320**, 122014.
- 182 Q. Yu, Y. Fu, J. Zhao, B. Li, X. Wang, X. Liu and L. Wang, *Appl. Catal., B*, 2023, **324**, 122297.
- 183 Z. Huang, S. Wang, R. D. Dewhurst, N. V. Ignat'ev, M. Finze and H. Braunschweig, *Angew. Chem., Int. Ed.*, 2020, **59**, 8800–8816.
- 184 P. Joshi, H.-H. Huang, R. Yadav, M. Hara and M. Yoshimura, *Catal. Sci. Technol.*, 2020, **10**, 6599–6610.
- 185 Y. Wen, J. Qi, D. Zhao, J. Liu, P. Wei, X. Kang and X. Li, *Appl. Catal., B*, 2021, **293**, 120196.
- 186 J. Xie, J. Zhang, S. Li, F. Grote, X. Zhang, H. Zhang, R. Wang, Y. Lei, B. Pan and Y. Xie, *J. Am. Chem. Soc.*, 2013, **135**, 17881–17888.
- 187 P. V. Sarma, C. S. Tiwary, S. Radhakrishnan, P. M. Ajayan and M. M. Shaijumon, *Nanoscale*, 2018, **10**, 9516–9524.
- 188 J. Xie, J. Xin, G. Cui, X. Zhang, L. Zhou, Y. Wang, W. Liu, C. Wang, M. Ning and X. Xia, *Inorg. Chem. Front.*, 2016, **3**, 1160–1166.
- 189 C. Wang, Z. Yu, H. Yao, R. Jin, K. Shi, Y. Li and S. Ma, *J. Alloys Compd.*, 2023, **964**, 171257.
- 190 G. Meng, Z. Chang, L. Zhu, C. Chen, Y. Chen, H. Tian, W. Luo, W. Sun, X. Cui and J. Shi, *Nano-Micro Lett.*, 2023, **15**, 212.
- 191 J. Xu, B. Yu, H. Zhao, S. Cao, L. Song, K. Xing, R. Zhou and X. Lu, *ACS Sustainable Chem. Eng.*, 2020, **8**, 15055–15064.
- 192 T. L. L. Doan, D. T. Tran, D. C. Nguyen, H. T. Le, N. H. Kim and J. H. Lee, *Appl. Catal., B*, 2020, **261**, 118268.
- 193 C. Liu, G. Zhang, L. Yu, J. Qu and H. Liu, *Small*, 2018, **14**, 1800421.
- 194 U. K. Sultana, T. He, A. Du and A. P. O'Mullane, *RSC Adv.*, 2017, **7**, 54995–55004.
- 195 J. Cui, S. Liang, X. Wang and J. Zhang, *Mater. Chem. Phys.*, 2015, **161**, 194–200.
- 196 W. Shi and Z. Wang, *J. Taiwan Inst. Chem. Eng.*, 2018, **82**, 163–168.
- 197 J. Lu, Z. Tang, L. Luo, S. Yin, P. K. Shen and P. Tsiakaras, *Appl. Catal., B*, 2019, **255**, 117737.
- 198 Q. Che, N. Bai, Q. Li, X. Chen, Y. Tan and X. Xu, *Nanoscale*, 2018, **10**, 15238–15248.
- 199 M. A. Ashraf, Y. Yang, D. Zhang and B. T. Pham, *J. Colloid Interface Sci.*, 2020, **577**, 265–278.
- 200 X. Fang, Z. Wang, Z. Jiang, J. Wang and M. Dong, *Electrochim. Acta*, 2019, **322**, 134739.
- 201 X. Sun, X. Gao, J. Chen, X. Wang, H. Chang, B. Li, D. Song, J. Li, H. Li and N. Wang, *ACS Appl. Mater. Interfaces*, 2020, **12**, 48591–48597.
- 202 Y. Tian, Z. Wei, X. Wang, S. Peng, X. Zhang and W. Liu, *Int. J. Hydrogen Energy*, 2017, **42**, 4184–4192.
- 203 X. Cui, Z. Chen, Z. Wang, M. Chen, X. Guo and Z. Zhao, *ACS Appl. Energy Mater.*, 2018, **1**, 5822–5829.
- 204 X. Zhang, S. Wang, C. Wu, H. Li, Y. Cao, S. Li and H. Xia, *J. Mater. Chem. A*, 2020, **8**, 23906–23918.
- 205 T. N. Kumar, N. Chandrasekaran and K. L. Phani, *Chem. Commun.*, 2015, **51**, 5052–5055.
- 206 N. Xue, Z. Lin, P. Li, P. Diao and Q. Zhang, *ACS Appl. Mater. Interfaces*, 2020, **12**, 28288–28297.
- 207 F. Liu, W. He, Y. Li, F. Wang, J. Zhang, X. Xu, Y. Xue, C. Tang, H. Liu and J. Zhang, *Appl. Surf. Sci.*, 2021, **546**, 149101.
- 208 Y.-N. Zhou, W.-L. Yu, Y.-N. Cao, J. Zhao, B. Dong, Y. Ma, F.-L. Wang, R.-Y. Fan, Y.-L. Zhou and Y.-M. Chai, *Appl. Catal., B*, 2021, **292**, 120150.
- 209 U. Sim, T.-Y. Yang, J. Moon, J. An, J. Hwang, J.-H. Seo, J. Lee, K. Y. Kim, J. Lee and S. Han, *Energy Environ. Sci.*, 2013, **6**, 3658–3664.
- 210 C. Lv, Q. Yang, Q. Huang, Z. Huang, H. Xia and C. Zhang, *J. Mater. Chem. A*, 2016, **4**, 13336–13343.
- 211 X. Wang, H. Huang, J. Qian, Y. Li and K. Shen, *Appl. Catal., B*, 2023, **325**, 122295.
- 212 X. Xu, K. Guo, J. Sun, X. Yu, X. Miao, W. Lu and L. Jiao, *Adv. Funct. Mater.*, 2024, **34**, 2400397.

