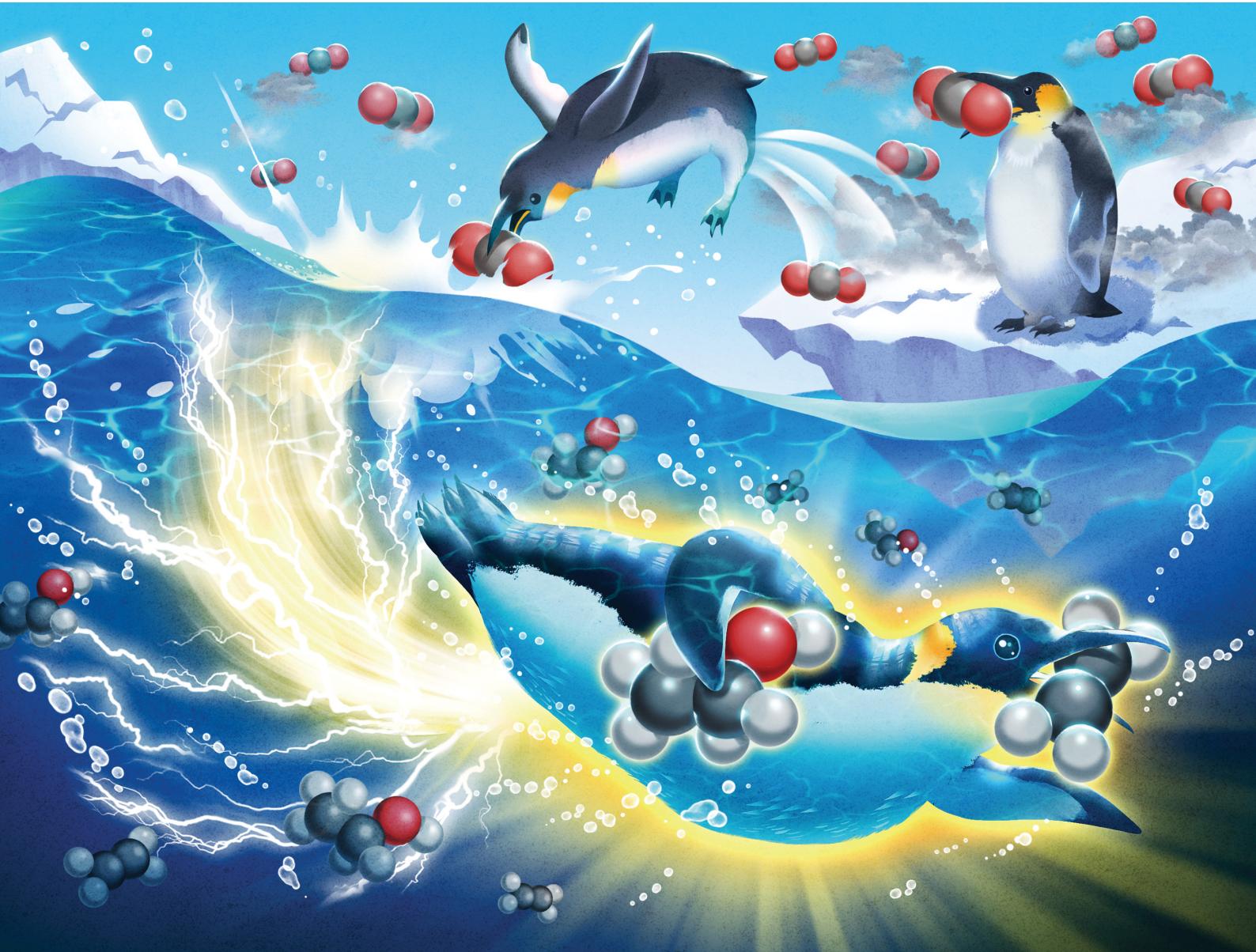


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## COMMUNICATION

Shuji Nakanishi, Kazuhide Kamiya *et al.*

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## Ultra-high-rate CO<sub>2</sub> reduction reactions to multicarbon products with a current density of 1.7 A cm<sup>-2</sup> in neutral electrolytes†

Asato Inoue,<sup>a</sup> Takashi Harada,<sup>ID ab</sup> Shuji Nakanishi<sup>\*ab</sup> and Kazuhide Kamiya<sup>ID \*ab</sup>

**CO<sub>2</sub> electrolysis to value-added products is a promising technology to close the carbon cycle and sequester anthropogenic CO<sub>2</sub> into chemical feedstocks; an increase of the current density for multicarbon products is one of the requirements for practical implementation. We have successfully increased the partial current density for gaseous CO<sub>2</sub> reduction reactions to multicarbon products (C<sub>2+</sub>) over Cu nanoparticles on gas diffusion electrodes in neutral electrolytes to a record value of 1.7 A cm<sup>-2</sup>. The faradaic efficiency for multicarbon products increased with the current density below total current density of 2000 mA cm<sup>-2</sup> and reached 76% at a total current density of 1600 mA cm<sup>-2</sup>. The turnover frequency for the production of C<sub>2+</sub> per Cu atoms exceeded 1.1 s<sup>-1</sup>. Optimizing the standard components and their assembly as the cathode elicits the high-turnover frequency of oxide-derived Cu catalysts, resulting in the record partial current density for C<sub>2+</sub>. Especially, we demonstrated that the thickness of catalyst layers was one highly sensitive factor in determining the maximum current density for C<sub>2+</sub>.**

### Broader context

The electrochemical reduction of carbon dioxide (CO<sub>2</sub>) has attracted considerable attention to maintain the carbon cycle. An increase in the current density for value-added multicarbon products (C<sub>2+</sub>), such as ethylene and ethanol, is required for practical implementation of this technology. Herein, we successfully obtained the partial current density for C<sub>2+</sub> over cupric oxide (CuO) nanoparticles on gas diffusion electrodes in neutral electrolytes to a value of 1.7 A cm<sup>-2</sup>. Additionally, the faradaic efficiency for multicarbon products increased with the current density below  $J_{\text{total}} = 2000$  mA cm<sup>-2</sup> and reached up to 76%. Our ultra-high-rate CO<sub>2</sub> electrolysis was achieved by optimizing the ordinary components, including oxide-derived Cu nanoparticles and carbon-based gas-diffusion electrodes, and their assembly as the cathode. In the other word, the present study succeeded in eliciting the potential performance of an ordinary combination. Although more detailed analyses are needed, our electrodes will teach us essential requirements for ultra-high-rate electrolysis. Additionally, the obtained knowledge through the detailed analyses of our electrodes would be widely utilized to develop novel materials, such as catalysts and electrodes.

## Introduction

The electrochemical carbon dioxide reduction reaction (CO<sub>2</sub>RR) has attracted considerable attention as a promising strategy for the conversion of anthropogenic CO<sub>2</sub> into value-added products.<sup>1–10</sup> One advantage of this method is that this greenhouse gas can be reduced under mild conditions (close to ambient temperature and pressure).<sup>11</sup> However, the operative efficiency, product selectivity and rate must be improved for practical implementation.<sup>12,13</sup> Among them, it has been reported

that the current density for the production of high value-added products is correlated to the capital cost of the electrodes employed.<sup>14</sup> Therefore, improvement of the current density would significantly affect the feasibility of this technology.

The use of gas diffusion electrodes (GDEs) that allow the CO<sub>2</sub>RR to occur at the solid-catalyst/liquid-electrolyte/gaseous-CO<sub>2</sub> interface, effectively accelerate the CO<sub>2</sub>RR by solving the problem of the mass transport limitation due to the inherently low diffusion and solubility of CO<sub>2</sub> in water.<sup>15,16</sup> A catalyst layer (CL) composed of metal Cu particles and ionomers on carbon-based microporous layers has been widely studied as the standard cathode for gaseous CO<sub>2</sub>RR to produce C<sub>2+</sub> organics, such as C<sub>2</sub>H<sub>4</sub>, C<sub>2</sub>H<sub>5</sub>OH, C<sub>3</sub>H<sub>7</sub>OH and CH<sub>3</sub>COOH.<sup>8,17–19</sup> Although the partial current density for C<sub>2+</sub> ( $J_{C_{2+}}$ ) has still remained below 500 mA cm<sup>-2</sup> in many studies,<sup>18,20–27</sup> a few studies have developed novel cathode assemblies or components toward an ultra-high-rate (UHR-) CO<sub>2</sub>RR with over  $J_{C_{2+}} = 1$  A cm<sup>-2</sup>. Sargent *et al.* successfully achieved a  $J_{C_{2+}}$  over 1.2 A cm<sup>-2</sup> in 7 M KOH using an electrode fabricated by

<sup>a</sup> Research Center for Solar Energy Chemistry, Graduate School of Engineering Science, Osaka University, 1-3 Machikaneyama, Toyonaka, Osaka 560-8531, Japan. E-mail: [nakanishi.shuji.es@osaka-u.ac.jp](mailto:nakanishi.shuji.es@osaka-u.ac.jp), [kamiya.kazuhide.es@osaka-u.ac.jp](mailto:kamiya.kazuhide.es@osaka-u.ac.jp)

<sup>b</sup> Innovative Catalysis Science Division, Institute for Open and Transdisciplinary Research Initiatives (ICS-OTRI), Osaka University, Suita, Osaka 565-0871, Japan

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dropping Cu nanoparticles on a sputtered Cu film/porous polytetrafluoroethylene (PTFE).<sup>28</sup> Additionally, in this work, the bulk heterojunction of catalysts and ionomers decoupled the pathways for gases, electrons and ions, which resulted in a  $j_{C_{2+}}$  over  $1 \text{ A cm}^{-2}$ . Wang and co-authors reported that a fluorine-modified copper catalyst deposited on GDEs exhibited a total current density ( $J_{\text{total}}$ ) of  $1600 \text{ mA cm}^{-2}$  with a faradaic efficiency for  $C_{2+}$  ( $FE_{C_{2+}}$ ) of over 80% (*i.e.*,  $j_{C_{2+}} = \text{over } 1.2 \text{ A cm}^{-2}$ ) in  $0.75 \text{ M KOH}$ .<sup>29</sup> Very recently, Zheng *et al.* obtained  $j_{C_{2+}} = 0.91 \text{ A cm}^{-2}$  using nitrogen-engineered Cu nanoparticles in  $1 \text{ M KOH}$ .<sup>10</sup>

In these reports, UHR-CO<sub>2</sub>RR has been achieved using high alkaline solutions to suppress H<sub>2</sub> evolution, which competes with the CO<sub>2</sub>RR. However, in addition to the toxicity and corrosivity, alkaline solutions quickly absorb CO<sub>2</sub> and convert it to electrochemically inert (bi)carbonate.<sup>30-33</sup> Based on these drawbacks, UHR electrolysis in neutral electrolytes is thus required from a practical viewpoint. Furthermore, these previous studies succeeded in achieving UHR-CO<sub>2</sub>RR by developing novel components in the three-phase interfaces (*i.e.*, catalysts or electrodes). However, the obtainable maximum  $j_{C_{2+}}$  value is still unknown when optimizing the standard components; CuO nanoparticle catalysts (CuONPs), which are known to serve as oxide-derived Cu catalysts for effective C<sub>2+</sub> production,<sup>34-36</sup> and carbon-based gas diffusion layers. An electrochemical system that can elicit the maximum  $j_{C_{2+}}$  using only standard components will become the primary platform for future studies to design materials and interfaces for UHR-CO<sub>2</sub>RR. In other words, the design guidelines obtained from the UHR-CO<sub>2</sub>RR with only ordinary materials would be quite general, and thus, they are expected to be widely applicable to novel materials and systems.

In the present work, we attempt to pursue a high partial current density for C<sub>2+</sub> ( $j_{C_{2+}}$ ) using neutral electrolytes by optimizing the standard components and properly assembling them as the cathode. In particular, we achieved a record current density of  $1.7 \text{ A cm}^{-2}$  for multicarbon products in neutral electrolytes using synthetic or commercial CuONPs and carbon-based gas diffusion layers. Firstly, we will explain the details in characterizations and CO<sub>2</sub>RR activity of our optimized electrode, and let us discuss the mechanism of UHR-CO<sub>2</sub>RR. Then, the requirements for our UHR electrolysis will be discussed with varying components and conditions.

## Results and discussion

### UHR-CO<sub>2</sub>RR over $1.7 \text{ A cm}^{-2}$ for C<sub>2+</sub> products obtained with CuONPs

In this first section, let us show the detailed characterizations and electrochemical properties of our optimized electrode for UHR-CO<sub>2</sub>RR, which is composed of the GDE (GDL 34BC) carrying our synthesized CuONPs. CuONP catalysts were synthesized by wet-chemical reduction using NaBH<sub>4</sub> as a reductant. Transmission electron microscopy (TEM) observations (Fig. 1a and Fig. S1, ESI<sup>†</sup>) show that the particle morphology and size is spherical and  $20 \text{ nm}$ , respectively. X-ray diffraction (XRD)

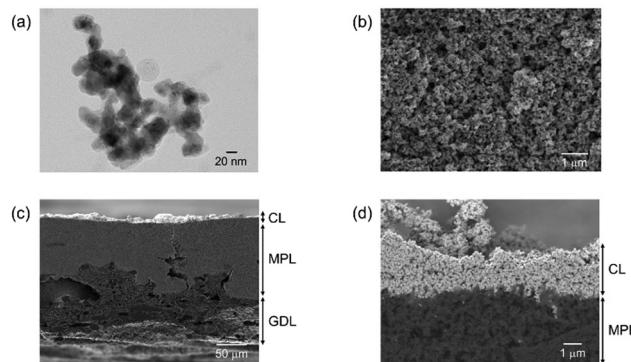


Fig. 1 (a) Representative TEM image of the synthesized CuONPs. (b) Top-view and (c and d) cross-sectional SEM micrographs of CuONPs-1.7/GDE.

patterns (Fig. S2, ESI<sup>†</sup>) indicate peaks at  $35.5^\circ$  and  $38.7^\circ$ , which are attributed to CuO(002) and CuO(111), respectively. This oxidation state of the as-prepared sample was also confirmed by narrow scan X-ray photoelectron spectroscopy (XPS) and X-ray absorption near edge structure (XANES), as shown in Fig. S3 and S4 (ESI<sup>†</sup>), respectively. Note that the Cu(II) oxidation state was reduced to Cu(I)/Cu(0) during electrolysis (Fig. S5 and S6, ESI<sup>†</sup>). Morphological characterization of the CuONPs/GDEs (Fig. S7, ESI<sup>†</sup>) was conducted using scanning electron microscopy (SEM). Fig. 1b-d and Fig. S8 (ESI<sup>†</sup>) show top- and cross-sectional SEM images of CuONPs-1.7/GDE (where the loading amount of CuONPs is  $1.7 \text{ mg cm}^{-2}$  on GDL 34BC), respectively. CuONPs completely covered the electrode surface based on the top-view SEM images, and were deposited on the microporous layer (MPL) composed of carbon nanoparticles. The cross-sectional SEM images reveal that the thickness of the MPL and the CL were *ca.* 80 and  $2 \mu\text{m}$  for CuONPs-1.7/GDE, respectively. When the loading amount of CuONPs was varied to  $0.34 \text{ mg cm}^{-2}$  (Fig. S9, ESI<sup>†</sup>) and  $3.1 \text{ mg cm}^{-2}$  (Fig. S10, ESI<sup>†</sup>), the CL thicknesses changed to *ca.* 0.5  $\mu\text{m}$  and 4  $\mu\text{m}$ , respectively. The roughness factors of the CL were obtained using a surface profilometer. When increasing CL thickness, the surface becomes rougher (Fig. S11 and Table S1, ESI<sup>†</sup>).

We measured the CO<sub>2</sub>RR activity of the Cu-based GDE catalysts. An in-house-built three-compartment electrochemical cell was used for CO<sub>2</sub>RR evaluation (Fig. S12, ESI<sup>†</sup>).<sup>37,38</sup>  $J_{\text{total}}$  vs. potential ( $U$ ) curves were obtained in  $1 \text{ M KCl}$  under CO<sub>2</sub> and argon conditions, as shown in Fig. S13 (ESI<sup>†</sup>). The onset potential of the cathodic current for bare and Cu-modified GDEs were  $-1.6$  and  $-1.2 \text{ V vs. Ag/AgCl}$  under CO<sub>2</sub>, respectively. When the Cu amount was increased from CuONPs-0.34/GDE to CuONPs-3.1/GDE, the cathodic current under CO<sub>2</sub> gradually increased.

The CO<sub>2</sub>RR products in outlet gas streams from the gas chamber and electrolytes from the catholyte room were analysed using gas chromatography (Fig. S14, ESI<sup>†</sup>) and <sup>1</sup>H nuclear magnetic resonance (NMR) spectroscopy (Fig. S15, ESI<sup>†</sup>) after constant-current electrolysis, respectively. Fig. 2a, b and Fig. S16 (ESI<sup>†</sup>) show the FE and partial current density of CO<sub>2</sub>RR products against the applied potential for CuONPs-1.7/GDE in  $1 \text{ M KCl}$ , respectively. Under low current conditions ( $J_{\text{total}} = 200 \text{ mA cm}^{-2}$ )



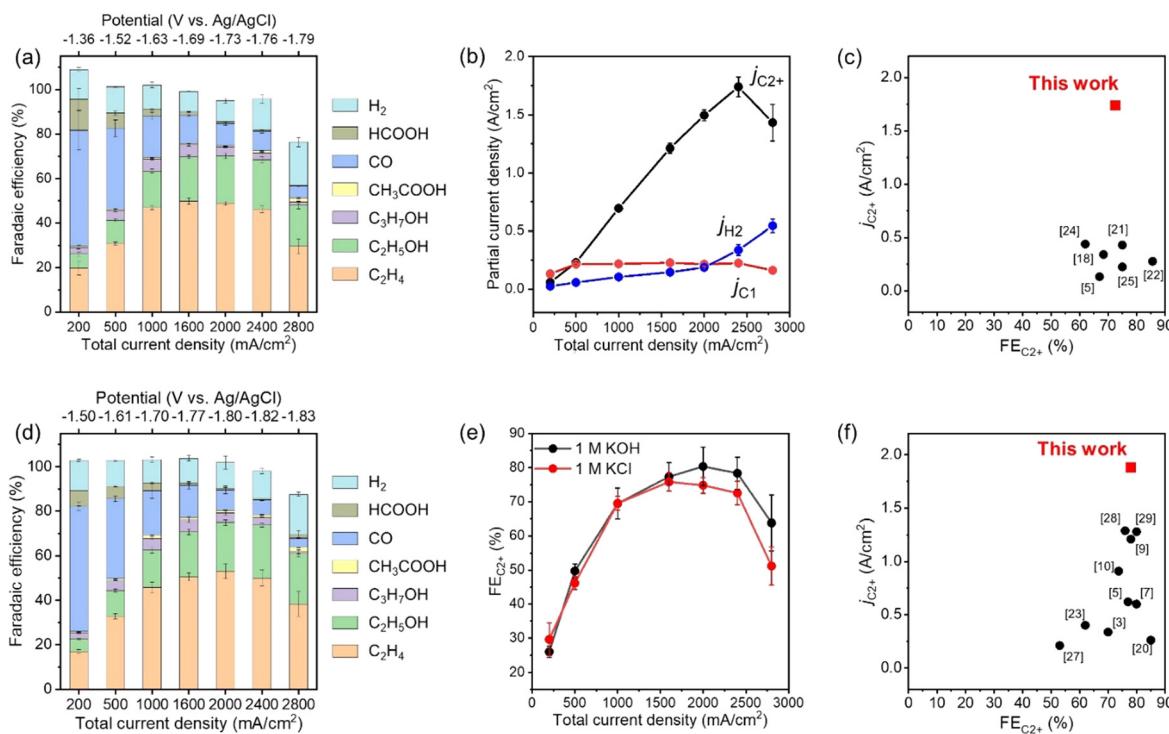


Fig. 2 (a) FEs of the  $\text{CO}_2$ RR products under various applied potentials on CuONPs-1.7/GDE in 1 M KCl. Reaction time: 30 min. (b) Partial current density for  $\text{C}_{2+}$ ,  $\text{C}_1$  and  $\text{H}_2$  on CuONPs-1.7/GDE in 1 M KCl. (c) Comparison of  $\text{FE}_{\text{C}_{2+}}$  and  $j_{\text{C}_{2+}}$  for CuONPs-1.7/GDE with reported Cu-based catalysts in neutral solution (for the detailed references, see Table S2, ESI<sup>†</sup>). (d) FEs for the  $\text{CO}_2$ RR products under various applied potentials on CuONPs-1.7/GDE in 1 M KOH. Reaction time: 30 min. (e) Comparison of  $\text{FE}_{\text{C}_{2+}}$  in 1 M KOH and in 1 M KCl under various  $J_{\text{total}}$  over CuONPs-1.7/GDE. (f) Comparison of  $\text{FE}_{\text{C}_{2+}}$  and  $j_{\text{C}_{2+}}$  for CuONPs-1.7/GDE with reported Cu-based catalysts in alkaline solution (for the detailed references, see Table S5, ESI<sup>†</sup>).

at  $-1.36 \text{ V vs. Ag/AgCl}$  electrode, CO was the main product ( $\text{FE}_{\text{CO}} > 50\%$ ), and the  $\text{FE}_{\text{C}_2\text{H}_4}$  was below 20%. The FE for  $\text{C}_{2+}$  products increased and reached 76% at  $-1.69 \text{ V vs. Ag/AgCl}$  electrode ( $J_{\text{total}} = 1600 \text{ mA cm}^{-2}$ ) when the applied potential was shifted to the negative side. In particular, the FEs for  $\text{C}_2\text{H}_4$  and  $\text{C}_2\text{H}_5\text{OH}$  were 46% and 22% at  $-1.76 \text{ V vs. Ag/AgCl}$  electrode, respectively. The partial current density for  $\text{C}_{2+}$  products ( $\text{C}_2\text{H}_4$ ,  $\text{C}_2\text{H}_5\text{OH}$ ,  $\text{C}_3\text{H}_7\text{OH}$ , and  $\text{CH}_3\text{COOH}$ ) was over  $j_{\text{C}_{2+}} = 1.7 \text{ A cm}^{-2}$  at  $-1.76 \text{ V vs. Ag/AgCl}$  electrode. On the contrary, the FEs for  $\text{C}_1$  products (CO,  $\text{CH}_4$  and HCOOH) and  $\text{H}_2$  were 8.8 and 14% at  $-1.76 \text{ V vs. Ag/AgCl}$  electrode, respectively. Fig. 2c and Table S2 (ESI<sup>†</sup>) show a comparison of  $j_{\text{C}_{2+}}$  and  $\text{FE}_{\text{C}_{2+}}$  in neutral electrolytes among the reported values. The partial current density for  $\text{C}_{2+}$  production ( $j_{\text{C}_{2+}} = \text{over } 1.7 \text{ A cm}^{-2}$ ) is a record value, and the  $\text{FE}_{\text{C}_{2+}}$  is also one of the highest. Therefore, the present study succeeded in eliciting the potential performance of an ordinary combination for UHR- $\text{CO}_2$ RR (*i.e.*, CuO-derived Cu nanoparticles and carbon-based GDEs). Significantly, such a high  $j_{\text{C}_{2+}}$  value for our assembly was remarkably reproducible even using different batches of electrodes, as shown in Fig. S17 and Table S3 (ESI<sup>†</sup>). When the applied potential was changed to  $-1.79 \text{ V vs. Ag/AgCl}$  electrode ( $J_{\text{total}} = 2800 \text{ mA cm}^{-2}$ ), the FE for  $\text{C}_{2+}$  products suddenly decreased, and the FE for  $\text{H}_2$  increased. The total FEs at  $-1.79 \text{ V vs. Ag/AgCl}$  electrode did not reach 100% (*ca.* 80%) because the bubbles contained  $\text{CO}_2$  substrate,  $\text{CO}_2$ RR products, and  $\text{H}_2$  formed at the cathode surface and diffused into the catholyte chamber. The applied potential was compensated

using the current interrupt method as written in the experimental section<sup>39–41</sup> (see Fig. S18 for the values *versus* reversible hydrogen electrode, ESI<sup>†</sup>). However, as accurate potential corrections are difficult under ultra-high-current conditions, we must note that the applied potentials are still referenced values.

Even when using neutral solutions, the local pH at the surface of the catalyst increases significantly during UHR- $\text{CO}_2$ RR due to the consumption of protons with an increase in the  $\text{CO}_2$ RR current density. Our rough calculation and previous reports showed that the local pH becomes almost the same in neutral and alkaline electrolytes during high-current electrolysis (Fig. S19 and Table S4, ESI<sup>†</sup>). Therefore, the cathode reaction proceeded under alkaline environments for UHR- $\text{CO}_2$ RR even with neutral solutions; thus, the fabricated cathode was expected to be suitable for UHR- $\text{CO}_2$ RR even in bulk alkaline solutions. Fig. 2d shows the FEs of  $\text{CO}_2$ RR products against the applied potential for CuONPs-1.7/GDE in 1 M KOH alkaline electrolyte, and Fig. 2e shows a comparison of the FEs of  $\text{C}_{2+}$  products for CuONPs-1.7/GDE in 1 M KCl and 1 M KOH solutions. The FEs for  $\text{CO}_2$ RR activity in alkaline and neutral solutions were almost the same in the range of  $J_{\text{total}} = 200\text{--}2400 \text{ mA cm}^{-2}$ , which indicates that the local pH around the CL in 1 M KOH and 1 M KCl solutions was not dependent on the bulk pH during electrolysis at  $J_{\text{total}} = 200 \text{ mA cm}^{-2}$ . The partial current density for  $\text{C}_{2+}$  products in alkaline solutions reached  $j_{\text{C}_{2+}} = 1.8 \text{ A cm}^{-2}$ , which is a record value in alkaline solution, as compared and listed in Fig. 2f and Table S5 (ESI<sup>†</sup>)

respectively, *i.e.*, this electrode can exhibit a record  $j_{C_{2+}}$  in both neutral and alkaline solutions. The mass activity, turnover frequency (TOF) and energy efficiency values are also summarized in Fig. S20 and Table S6 (ESI<sup>†</sup>) as reference data.

The volcano-type relation between  $FE_{C_{2+}}$  and current density was observed in the present UHR-CO<sub>2</sub>RR system. Especially, the  $FE_{C_{2+}}$  increased with increasing current density below  $J_{\text{total}} = 2000 \text{ mA cm}^{-2}$ , and thus, a trade-on relationship between rate and selectivity was exhibited in this current region, as shown in Fig. 2a. Let us discuss the mechanism for the increase of  $FE_{C_{2+}}$  with the current density. It should be noted here that it is difficult to strictly distinguish whether the selectivity toward the product depends on the current density (*i.e.*, reaction rate) or on the applied potential because we cannot control each parameter independently. We here explain the product selectivity from the viewpoint of current density: the effect of (1) CO partial pressure, (2) local pH,<sup>42</sup> and (3) *in situ* formation of the three-phase interface. The CO partial pressure in the CL became high with an increase in the current density, which facilitated CO dimerization. Jun Li *et al.* controlled local CO availability at the catalyst–electrolyte interface and demonstrated that a high concentration of CO accelerated CO dimerization, and caused a higher current density with  $C_{2+}$  production.<sup>43</sup> Jing Li and co-workers reported that a high CO partial pressure suppresses  $H^*$  adsorption, which leads to lowering of the hydrogen evolution reaction (HER).<sup>44</sup> It should also be noted here that the FE for  $C_2H_5OH$  against  $C_2H_4$  increased with increasing  $J_{\text{total}}$  (Fig. S21, ESI<sup>†</sup>). It has been reported that high CO pressure on a Cu surface favours  $C_2H_5OH$  production over  $C_2H_4$  because the  $^*CHCHOH$  intermediate for  $C_2H_5OH$  becomes more stable than  $^*CCH$  for  $C_2H_4$  under high CO coverage.<sup>43,45</sup> This is because  $CCH^*$  and  $^*CHCHOH$  are adsorbed on Cu sites with side-on and end-on configurations, respectively, so that  $CCH^*$  requires a larger occupied molecular area than  $^*CHCHOH$ . This change in selectivity between  $C_2H_4$  and  $C_2H_5OH$  supports that the CO local pressure depended on the current density. The increase in the local pH at the surface of the CL due to proton consumption by the CO<sub>2</sub>RR suppressed HER (*vide supra*). The pH at the electrode surfaces is known to reach up to 11.9 under electrolysis at just  $J_{\text{total}} = 50 \text{ mA cm}^{-2}$  in neutral electrolytes.<sup>15,46</sup> Considering that the almost same product selectivity was obtained in neutral (1 M KCl) and alkaline (1 M KOH) solutions between  $J_{\text{total}} = 200\text{--}2400 \text{ mA cm}^{-2}$ , the local pH in KCl solution became similar to that in KOH solution under electrolysis with  $J_{\text{total}}$  of just  $J_{\text{total}} = 200 \text{ mA cm}^{-2}$ . Further increase in the current density ( $J_{\text{total}} > 1000 \text{ mA cm}^{-2}$ ) would lead to a pH of over 15 at the surface of the catalyst.<sup>7</sup> Moreover, we propose that the *in situ* formation of the gas-transport pathway depending on the current density occurs during electrolysis. Evolved gases, such as  $H_2$ , would migrate through gas-transport channels in the CL; therefore, the area of the CO<sub>2</sub>RR active interface (three-phase interfaces) could be expanded with an increase in the current density (see Fig. S22, ESI<sup>†</sup>).

It is important to identify the origin of the decrease in the CO<sub>2</sub>RR activity by application of  $J_{\text{total}} = 2800 \text{ mA cm}^{-2}$  for

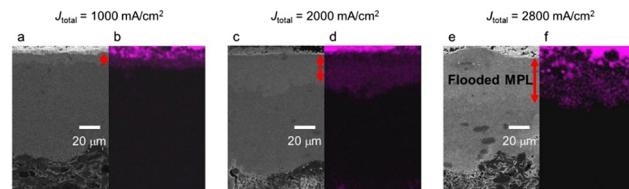


Fig. 3 (a, c and e) Cross-sectional SEM images of CuONPs-1.7/GDE and (b, d and f)  $K^+$  distribution (purple) obtained by EDX after application of  $J_{\text{total}} =$  (a and b)  $1000 \text{ mA cm}^{-2}$ , (c and d)  $2000 \text{ mA cm}^{-2}$ , and (e and f)  $2800 \text{ mA cm}^{-2}$  for 30 min under CO<sub>2</sub>RR conditions in 1 M KCl.

CuONPs-1.7/GDE (Fig. 2a and d) toward a further increase in the production rate. It was assumed that the three-phase interface was broken over  $J_{\text{total}} = 2800 \text{ mA cm}^{-2}$ , which resulted in a severe decrease in the FE for  $C_{2+}$  products. Fig. 3 shows cross-sectional SEM images after application of  $J_{\text{total}} = 1000$ , 2000, and  $2800 \text{ mA cm}^{-2}$  (30 min) in 1 M KCl, respectively. The particle size of the CuONPs became about 10% smaller after electrolysis due to the reduction of CuONPs to Cu(*l*)/Cu(*0*)NPs (Fig. S23, ESI<sup>†</sup>). Cross-sectional elemental energy dispersive X-ray spectroscopy (EDX) mapping show  $K^+$  derived from electrolytes entered the MPL layer. One-tenth and one-fourth of the MPL layers were flooded after application of  $J_{\text{total}} = 1000 \text{ mA cm}^{-2}$  and  $2000 \text{ mA cm}^{-2}$ , respectively. In contrast, after the degradation with application of  $J_{\text{total}} = 2800 \text{ mA cm}^{-2}$  (30 min), the  $K^+$  distribution was shown to reach half of the MPL layer (Fig. 3e). In addition, a residue of  $K^+$  crystals ( $KHCO_3$  and  $K_2CO_3$ , KCl and KOH) was observed throughout the MPL after application of  $J_{\text{total}} = 2800 \text{ mA cm}^{-2}$  (Fig. S24, ESI<sup>†</sup>). Electrowetting effects, which decrease the solid–liquid interfacial tension and facilitate electrolyte penetration into the CL and MPL, are known to be emphasized under application of highly negative potentials.<sup>47–49</sup> Although the evolved gases may push out and expand the gas-transport pathways (as the *in situ* formation of the three-phase interface), flooding of the MPL due to electrowetting may exceed the rate of pushing out the electrolyte. The flooded CL (and MPL) produced only  $H_2$  due to a lack of CO<sub>2</sub> transportation, and thus,  $H_2$  production should become dominant at  $J_{\text{total}} = 2800 \text{ mA cm}^{-2}$ . This assumption is also supported by the diffusion of evolved gases into the catholyte chamber at  $J_{\text{total}} = 2800 \text{ mA cm}^{-2}$  (*vide supra*).

#### UHR-CO<sub>2</sub>RR with varying components and conditions

Here, let us attempt to reveal which factors are sensitive for our UHR-CO<sub>2</sub>RR by varying several components and conditions.

Fig. S25 and Table S7 (ESI<sup>†</sup>) show the FE of CO<sub>2</sub>RR products under  $J_{\text{total}} = 2000 \text{ mA cm}^{-2}$  when changing catalysts, the thickness of CL, electrolytes, and GDEs. One can notice that several kinds of commercial GDEs are available for UHR-CO<sub>2</sub>RR. Additionally, the high  $j_{C_{2+}}$  of over  $1.3 \text{ A cm}^{-2}$  was obtained using 1 M  $KHCO_3$ , which has been widely used as a neutral electrolyte for CO<sub>2</sub>RR. The slight decrease of  $j_{C_{2+}}$  in 1 M  $KHCO_3$  is likely due to an acceleration of salt precipitation in MPL in the bicarbonate-rich electrolytes.

Surprisingly, UHR-CO<sub>2</sub>RR with  $j_{C_{2+}}$  of over  $1.4 \text{ A cm}^{-2}$  was realized even when using commercial CuONPs with the size of



50 nm. These results indicated that the GDE and catalyst used in this study are not unique, and the optimized assembly elicits their undermined potential. Further studies are currently in progress to determine the detailed requirements for catalysts in our laboratory.

Among these parameters, Fig. S25 (ESI†) reveals that the thickness of CL is one high-sensitivity factor in determining maximum  $j_{C_{2+}}$ . The optimal thickness of CL was required for UHR-CO<sub>2</sub>RR, not too thin and not too thick. Thus, we investigated the detail of the dependence of CL thickness. The CO<sub>2</sub>RR was investigated for CuONPs-3.1/GDE and CuONPs-0.34/GDE (detailed product selectivities are shown in Fig. S26 and S27, ESI†) and the standard CuONPs-1.7/GDE (see the FEs in Fig. 2a) in 1 M KCl. The maximum  $j_{C_{2+}}$  for CuONPs-3.1/GDE was achieved at  $J_{\text{total}} = 2400 \text{ mA cm}^{-2}$ , which was the same as that for CuONPs-1.7/GDE. The FEs for C<sub>2+</sub> over CuONPs-3.1/GDE were 7% lower than those over CuONPs-1.7/GDE, although there was a similar tendency of the potential dependence (Fig. 4a). The FE<sub>H<sub>2</sub></sub> over CuONPs-3.1/GDE was higher than that over CuONPs-1.7/GDE for all current density regions (Fig. S28, ESI†). These results indicated that a lack of Cu active sites did not cause a decrease in FE<sub>C<sub>2+</sub></sub> over  $J_{\text{total}} = 2400 \text{ mA cm}^{-2}$  for CuONPs-1.7/GDE. It was presumed that the Cu sites immersed in the electrolyte cannot serve as CO<sub>2</sub>RR sites but only as HER sites, and thus, selectivity toward H<sub>2</sub> production increased with the amount of Cu (Fig. 4d). Next, the Cu amount on the GDE was decreased to 0.34 mg cm<sup>-2</sup>. Although the FE<sub>C<sub>2+</sub></sub> over CuONPs-0.34/GDE was greater than 80% at  $J_{\text{total}} = 1000 \text{ mA cm}^{-2}$ , it decreased to below 60% at  $J_{\text{total}} = 1600 \text{ mA cm}^{-2}$ . In contrast, when  $J_{\text{total}}$  was varied from 1000 mA cm<sup>-2</sup> to 1600 mA cm<sup>-2</sup>, the FEs for CH<sub>4</sub> and H<sub>2</sub> increased from 1.6% to 14%, and from 5.4%

to 17%, respectively. This is because the lack of catalytic Cu sites would result in relative enhancement of the HER on carbon particles that compose the MPL. In other words, the three-phase interface formed in CuONPs-0.34/GDE was thinner than that in CuONPs-1.7/GDE, resulting that the current density with the highest FE<sub>C<sub>2+</sub></sub> was smaller than the optimal for CuONPs-1.7/GDE (Fig. 4b and c). A significant increase in FE for CH<sub>4</sub> with an increase in current density was observed on CuONPs-0.34/GDE. This tendency for CH<sub>4</sub> was basically consistent with the reported potential dependence of the selectivity of the CO<sub>2</sub>RR on Cu-based catalysts.<sup>50–52</sup> The same trends in the catalytic activity under various amounts of catalyst loadings were observed in 1 M KOH electrolyte (Fig. S29–S31, ESI†).

Fig. S32 (ESI†) shows the FEs depending on CL thickness at 500, 1000, 1600 and 2000 mA cm<sup>-2</sup>, to further demonstrate the importance of CL thickness for maximum  $j_{C_{2+}}$ . The thin CL (CuONPs-0.34) showed the best FE for C<sub>2+</sub>, and FE<sub>C<sub>2+</sub></sub> monotonically decreased with increasing the thickness of CL at  $J_{\text{total}} = 500$  and 1000 mA cm<sup>-2</sup>. In contrast to FE<sub>C<sub>2+</sub></sub>, the total FE for CO<sub>2</sub>RR was almost constant (85–90%) at these low current density regions, even when varying CL thickness. This is likely because CO partial pressure was not enough to form C–C bonds for thicker CL at the low current density regions. When increasing current density to  $J_{\text{total}} = 1600 \text{ mA cm}^{-2}$ , the FE<sub>C<sub>2+</sub></sub> for CuONPs-1.7 became the best in these three electrodes. Thus, our results clearly indicated that an optimal CL thickness for C<sub>2+</sub> evolution depends on  $J_{\text{total}}$ .

Although our study is not the first paper to optimize and discuss each parameter, including the size of catalysts, porosity, and thickness of CL, we think that these interdependent parameters were optimized simultaneously, likely resulting in the highest  $j_{C_{2+}}$  in this work. However, we must note that some requirements, especially for catalysts, toward UHR-CO<sub>2</sub>RR are still veiled. This work is focused on demonstrating the potential of ordinary catalysts; providing more detailed requirements will be left for a future paper.

### Long-term stability dependence on current density

Long-term stability might be a challenging issue in the practical implementation of UHR-CO<sub>2</sub>RR. For example, in ref. 29 the authors reported that the FE<sub>C<sub>2+</sub></sub> started to decrease within one hour at  $J_{\text{total}} = 1200 \text{ mA cm}^{-2}$  ( $j_{C_{2+}} = 1 \text{ A cm}^{-2}$ ) in 1 M KOH. In this time scale, the decrease in the CO<sub>2</sub>RR activity on GDE is not due to the chemical degradation of components, but due to the flooding of the CL and MPL. However, there is no systematic knowledge about the long-term stability of UHR-CO<sub>2</sub>RR; especially, a durability test of UHR-CO<sub>2</sub>RR in neutral solutions has not been conducted at all. Therefore, we analysed the change in the CO<sub>2</sub>RR activity for long-term electrolysis under ultra-high-rate electrolysis conditions in neutral solutions.

Fig. 5b and Fig. S33 (ESI†) show the FEs for CO<sub>2</sub>RR products at  $J_{\text{total}} = 1600 \text{ mA cm}^{-2}$  in 1 M KCl as a function of the electrolysis time. The change in the FEs for CO<sub>2</sub>RR products under electrolysis at  $J_{\text{total}} = 1600 \text{ mA cm}^{-2}$  was almost negligible for over 2 h. Although the FE for C<sub>2+</sub> and H<sub>2</sub> then began to decrease and increase from 2 h, respectively, the FE<sub>C<sub>2+</sub></sub>

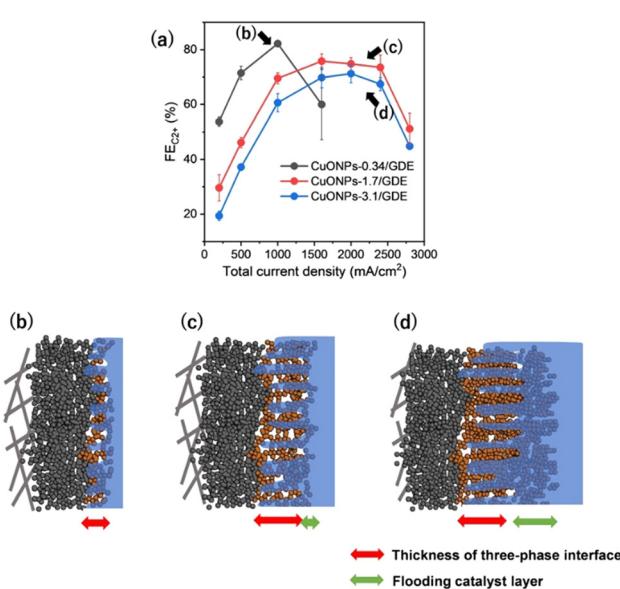


Fig. 4 (a) FE<sub>C<sub>2+</sub></sub> against  $J_{\text{total}}$  for various amounts of Cu loading (CuONPs-0.34, -1.7 and -3.1/GDE) in 1 M KCl. (b–d) Schematic image of the three-phase interfaces at each position which are indicated in (a). The red and green arrows represent the thickness of three-phase interface and flooding catalyst layer, respectively.



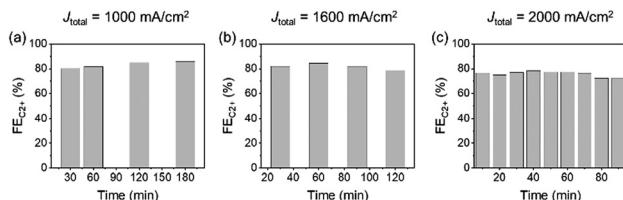


Fig. 5 FE<sub>C<sub>2+</sub></sub> stability tests with respect to the current density over CuONPs-1.0/GDE in 1 M KCl under (a)  $J_{\text{total}} = 1000 \text{ mA cm}^{-2}$ , (b)  $1600 \text{ mA cm}^{-2}$  and (c)  $2000 \text{ mA cm}^{-2}$ .

production remained over 75% for 2.5 h and 60% for over 6 h. The electrode was analysed after the decrease of FEs for C<sub>2+</sub> to clarify the origin of activity decrease. SEM observation (Fig. S34, ESI†) after electrolysis for 6 h at  $1600 \text{ mA cm}^{-2}$  showed that the morphology of CLs did not significantly vary, and most of the MPL was flooded in a similar manner to that described in previous reports.<sup>53,54</sup> The recycling test was also conducted. Fig. S35 (ESI†) shows that the FE<sub>C<sub>2+</sub></sub> value was recovered by washing the electrode with pure water after the decrease. This result also indicates that the decrease in FE<sub>C<sub>2+</sub></sub> was due not to the degradation of catalysts but to the salt precipitation induced by flooded MPL. The stability dependence on the applied current density was investigated next (Fig. 5a–c and Fig. S33, ESI†). The FE<sub>C<sub>2+</sub></sub> of 70–80% almost unvaried for over 3 h and 1.5 h at  $J_{\text{total}} = 1000 \text{ mA cm}^{-2}$  and  $J_{\text{total}} = 2000 \text{ mA cm}^{-2}$ , respectively. These results indicated a trade-off relationship between stability and current density; however, the decrease in FE<sub>C<sub>2+</sub></sub> within a short timescale was due to flooding of the CL and MPLs. Therefore, we can expect that the stability could be improved by the addition of hydrophobic materials and modulation of the porosity in the CL and MPL. The present electrochemical system and conditions will become one of the standard platforms toward these future studies.

## Conclusions

In this work, our electrode exhibited partial current densities for CO<sub>2</sub> reduction to C<sub>2+</sub> of  $1.7 \text{ A cm}^{-2}$  in neutral solutions by optimizing the standard components and their assembly. When the total current density was increased, the FEs for C<sub>2+</sub> increased monotonically below  $J_{\text{total}} = 2000 \text{ mA cm}^{-2}$ . Therefore, the trade-on relationship between the reaction rate and selectivity was shown in the present UHR-CO<sub>2</sub>RR under those  $J_{\text{total}}$  regions. Although the cathode of this electrochemical system was composed of typical CuO-derived Cu nanoparticles, a carbon-based GDE, we successfully maximized the potential of the catalytic activity of the Cu surface by constructing a CL with optimal porosity and thickness. Especially, we revealed that the thickness of CL was directly correlated to  $j_{C_{2+}}$ . The optimal thickness of CL was required for UHR-CO<sub>2</sub>RR. It is noteworthy that this electrode allowed us to achieve UHR-CO<sub>2</sub>RR even in neutral solutions, which is more practical than alkaline solutions. The present system is expected to become one of the standards when pursuing high-current density for the CO<sub>2</sub>RR by the design of novel components or assemblies. Although more detailed

analyses are needed, our electrodes will teach us essential requirements for UHR electrolysis. Furthermore, the obtained knowledge through the detailed analyses of our electrodes would be widely utilized to develop novel materials, such as catalysts and electrodes because our electrodes are composed of only standard components. Systematic stability tests were also conducted at over  $j_{C_{2+}} = 1 \text{ A cm}^{-2}$ . Although the stability under ultra-high-current conditions was still far from a practical level, SEM observations revealed that flooding of the CL and MPL is a problem that occurs within the shortest time scale. The electrochemical conditions and systems in this study can serve as a platform for improvement of the stability of the UHR-CO<sub>2</sub>RR.

## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 W. Zhang, Y. Hu, L. Ma, G. Zhu, Y. Wang, X. Xue, R. Chen, S. Yang and Z. Jin, *Adv. Sci.*, 2018, **5**, 1700275.
- 2 T. Zhang, J. C. Bui, Z. Li, A. T. Bell, A. Z. Weber and J. Wu, *Nat. Catal.*, 2022, **5**, 202–211.
- 3 G. Zhang, Z. J. Zhao, D. Cheng, H. Li, J. Yu, Q. Wang, H. Gao, J. Guo, H. Wang, G. A. Ozin, T. Wang and J. Gong, *Nat. Commun.*, 2021, **12**, 5745.
- 4 Z. W. She, J. Kibsgaard, C. F. Dickens, I. Chorkendorff, J. K. Nørskov and T. F. Jaramillo, *Science*, 2017, **355**, eaad4998.
- 5 T. H. M. Pham, J. Zhang, M. Li, T. H. Shen, Y. Ko, V. Tileli, W. Luo and A. Züttel, *Adv. Energy Mater.*, 2022, **12**, 2103663.
- 6 Y. Hori, H. Wakebe, T. Tsukamoto and O. Koga, *Electrochim. Acta*, 1994, **39**, 1833–1839.
- 7 C. T. Dinh, T. Burdyny, G. Kibria, A. Seifitokaldani, C. M. Gabardo, F. Pelayo García De Arquer, A. Kiani, J. P. Edwards, P. De Luna, O. S. Bushuyev, C. Zou, *Electrochim. Acta*, 2022, **434**, 139533.



R. Quintero-Bermudez, Y. Pang, D. Sinton and E. H. Sargent, *Science*, 2018, **360**, 783–787.

8 H. Li, T. Liu, P. Wei, L. Lin, D. Gao, G. Wang and X. Bao, *Angew. Chem., Int. Ed.*, 2021, **60**, 14329–14333.

9 C. Chen, X. Yan, Y. Wu, S. Liu, X. Zhang, X. Sun, Q. Zhu, H. Wu and B. Han, *Angew. Chem., Int. Ed.*, 2022, **61**, e2022026.

10 M. Zheng, P. Wang, X. Zhi, K. Yang, Y. Jiao, J. Duan, Y. Zheng and S.-Z. Qiao, *J. Am. Chem. Soc.*, 2022, **144**, 14936–14944.

11 K. Kamiya, K. Fujii, M. Sugiyama and S. Nakanishi, *Chem. Lett.*, 2021, **50**, 166–179.

12 P. De Luna, C. Hahn, D. Higgins, S. A. Jaffer, T. F. Jaramillo and E. H. Sargent, *Science*, 2019, **364**, eaav.3506.

13 J. M. Spurgeon and B. Kumar, *Energy Environ. Sci.*, 2018, **11**, 1536–1551.

14 S. Verma, B. Kim, H. R. M. Jhong, S. Ma and P. J. A. Kenis, *ChemSusChem*, 2016, **9**, 1972–1979.

15 T. Burdyny and W. A. Smith, *Energy Environ. Sci.*, 2019, **12**, 1442–1453.

16 Z. Xing, L. Hu, D. S. Ripatti, X. Hu and X. Feng, *Nat. Commun.*, 2021, **12**, 136.

17 S. Ma, M. Sadakiyo, R. Luo, M. Heima, M. Yamauchi and P. J. A. Kenis, *J. Power Sources*, 2016, **301**, 219–228.

18 Y. C. Tan, K. B. Lee, H. Song and J. Oh, *Joule*, 2020, **4**, 1104–1120.

19 P. P. Yang, X. L. Zhang, F. Y. Gao, Y. R. Zheng, Z. Z. Niu, X. Yu, R. Liu, Z. Z. Wu, S. Qin, L. P. Chi, Y. Duan, T. Ma, X. S. Zheng, J. F. Zhu, H. J. Wang, M. R. Gao and S. H. Yu, *J. Am. Chem. Soc.*, 2020, **142**, 6400–6408.

20 T. T. H. Hoang, S. Verma, S. Ma, T. T. Fister, J. Timoshenko, A. I. Frenkel, P. J. A. Kenis and A. A. Gewirth, *J. Am. Chem. Soc.*, 2018, **140**, 5791–5797.

21 M. G. Kibria, C. T. Dinh, A. Seifitokaldani, P. De Luna, T. Burdyny, R. Quintero-Bermudez, M. B. Ross, O. S. Bushuyev, F. P. García de Arquer, P. Yang, D. Sinton and E. H. Sargent, *Adv. Mater.*, 2018, **30**, 1804867.

22 F. Li, A. Thevenon, A. Rosas-Hernández, Z. Wang, Y. Li, C. M. Gabardo, A. Ozden, C. T. Dinh, J. Li, Y. Wang, J. P. Edwards, Y. Xu, C. McCallum, L. Tao, Z. Q. Liang, M. Luo, X. Wang, H. Li, C. P. O'Brien, C. S. Tan, D. H. Nam, R. Quintero-Bermudez, T. T. Zhuang, Y. C. Li, Z. Han, R. D. Britt, D. Sinton, T. Agapie, J. C. Peters and E. H. Sargent, *Nature*, 2020, **577**, 509–513.

23 J. J. Lv, M. Jouny, W. Luc, W. Zhu, J. J. Zhu and F. Jiao, *Adv. Mater.*, 2018, **30**, 1803111.

24 T. Möller, T. Ngo Thanh, X. Wang, W. Ju, Z. Jovanov and P. Strasser, *Energy Environ. Sci.*, 2021, **14**, 5995–6006.

25 Z. Z. Wu, X. L. Zhang, Z. Z. Niu, F. Y. Gao, P. P. Yang, L. P. Chi, L. Shi, W. S. Wei, R. Liu, Z. Chen, S. Hu, X. Zheng and M. R. Gao, *J. Am. Chem. Soc.*, 2022, **144**, 259–269.

26 X. Zhang, J. Li, Y. Y. Li, Y. Jung, Y. Kuang, G. Zhu, Y. Liang and H. Dai, *J. Am. Chem. Soc.*, 2021, **143**, 3245–3255.

27 T. T. Zhuang, Z. Q. Liang, A. Seifitokaldani, Y. Li, P. De Luna, T. Burdyny, F. Che, F. Meng, Y. Min, R. Quintero-Bermudez, C. T. Dinh, Y. Pang, M. Zhong, B. Zhang, J. Li, P. N. Chen, X. L. Zheng, H. Liang, W. N. Ge, B. J. Ye, D. Sinton, S. H. Yu and E. H. Sargent, *Nat. Catal.*, 2018, **1**, 421–428.

28 F. P. García de Arquer, C. T. Dinh, A. Ozden, J. Wicks, C. McCallum, A. R. Kirmani, D. H. Nam, C. Gabardo, A. Seifitokaldani, X. Wang, Y. C. Li, F. Li, J. Edwards, L. J. Richter, S. J. Thorpe, D. Sinton and E. H. Sargent, *Science*, 2020, **367**, 661–666.

29 W. Ma, S. Xie, T. Liu, Q. Fan, J. Ye, F. Sun, Z. Jiang, Q. Zhang, J. Cheng and Y. Wang, *Nat. Catal.*, 2020, **3**, 478–487.

30 M. Ma, E. L. Clark, K. T. Therkildsen, S. Dalsgaard, I. Chorkendorff and B. Seger, *Energy Environ. Sci.*, 2020, **13**, 977–985.

31 J. A. Rabinowitz and M. W. Kanan, *Nat. Commun.*, 2020, **11**, 5231.

32 R. Xia, J. J. Lv, X. Ma and F. Jiao, *J. Catal.*, 2021, **398**, 185–191.

33 J. E. Huang, F. Li, A. Ozden, A. S. Rasouli, F. P. G. de Arquer, S. Liu, S. Zhang, M. Luo, X. Wang, Y. Lum, Y. Xu, K. Bertens, R. K. Miao, C. T. Dinh, D. Sinton and E. H. Sargent, *Science*, 2021, **372**, 1074–1078.

34 D. Cheng, Z.-J. Zhao, G. Zhang, P. Yang, L. Li, H. Gao, S. Liu, X. Chang, S. Chen, T. Wang, G. A. Ozin, Z. Liu and J. Gong, *Nat. Commun.*, 2021, **12**, 395.

35 X. Wang, K. Klingan, M. Klingenhof, T. Möller, J. Ferreira de Araújo, I. Martens, A. Bagger, S. Jiang, J. Rossmeisl, H. Dau and P. Strasser, *Nat. Commun.*, 2021, **12**, 794.

36 W. Liu, P. Zhai, A. Li, B. Wei, K. Si, Y. Wei, X. Wang, G. Zhu, Q. Chen, X. Gu, R. Zhang, W. Zhou and Y. Gong, *Nat. Commun.*, 2022, **13**, 1877.

37 S. Kato, K. Iwase, T. Harada, S. Nakanishi and K. Kamiya, *ACS Appl. Mater. Interfaces*, 2020, **12**, 29376–29382.

38 Y. Wu, K. Iwase, T. Harada, S. Nakanishi and K. Kamiya, *ACS Appl. Nano Mater.*, 2021, **4**, 4994–5003.

39 A. R. Heenan, J. Hamonnet and A. T. Marshall, *ACS Energy Lett.*, 2022, **7**, 2357–2361.

40 K. Liu, W. A. Smith and T. Burdyny, *ACS Energy Lett.*, 2019, **4**, 639–643.

41 M. Hussainy and D. W. Agar, *Chem. Eng. Technol.*, 2016, **39**, 2135–2141.

42 J. Chen and L. Wang, *Adv. Mater.*, 2021, **34**, 2103900.

43 J. Li, Z. Wang, C. McCallum, Y. Xu, F. Li, Y. Wang, C. M. Gabardo, C. T. Dinh, T. T. Zhuang, L. Wang, J. Y. Howe, Y. Ren, E. H. Sargent and D. Sinton, *Nat. Catal.*, 2019, **2**, 1124–1131.

44 J. Li, K. Chang, H. Zhang, M. He, W. A. Goddard, J. G. Chen, M. J. Cheng and Q. Lu, *ACS Catal.*, 2019, **9**, 4709–4718.

45 F. Li, Y. C. Li, Z. Wang, J. Li, D. H. Nam, Y. Lum, M. Luo, X. Wang, A. Ozden, S. F. Hung, B. Chen, Y. Wang, J. Wicks, Y. Xu, Y. Li, C. M. Gabardo, C. T. Dinh, Y. Wang, T. T. Zhuang, D. Sinton and E. H. Sargent, *Nat. Catal.*, 2020, **3**, 75–82.

46 X. Lu, C. Zhu, Z. Wu, J. Xuan, J. S. Francisco and H. Wang, *J. Am. Chem. Soc.*, 2020, **142**, 15438–15444.

47 E. R. Cofell, U. O. Nwabara, S. S. Bhargava, D. E. Henckel and P. J. A. Kenis, *ACS Appl. Mater. Interfaces*, 2021, **13**, 15132–15142.

48 K. Junge Puring, D. Siegmund, J. Timm, F. Möllenbrück, S. Schemme, R. Marschall and U. P. Apfel, *Adv. Sustainable Syst.*, 2021, **5**, 2000088.



49 K. Yang, R. Kas, W. A. Smith and T. Burdyny, *ACS Energy Lett.*, 2021, **6**, 33–40.

50 Y. Huang, A. D. Handoko, P. Hirunsit and B. S. Yeo, *ACS Catal.*, 2017, **7**, 1749–1756.

51 K. P. Kuhl, E. R. Cave, D. N. Abram and T. F. Jaramillo, *Energy Environ. Sci.*, 2012, **5**, 7050–7059.

52 K. Manthiram, B. J. Beberwyck and A. P. Alivisatos, *J. Am. Chem. Soc.*, 2014, **136**, 13319–13325.

53 Y. Kong, H. Hu, M. Liu, Y. Hou, V. Kolivoška, S. Vesztergom and P. Broekmann, *J. Catal.*, 2022, **408**, 1–8.

54 Y. Wu, S. Garg, M. Li, M. N. Idros, Z. Li, R. Lin, J. Chen, G. Wang and T. E. Rufford, *J. Power Sources*, 2022, **522**, 230998.

