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Engineered interface coverage and precise cocatalyst placement in MOF-derived heterojunction photocatalysts for selective methane oxidation

Wendi Zhao,^{†a} Kang Sun,^{†a} Jiayi Xu,^a Zhongyuan Lin,^a Qihui Chen,^{ID *b} Maochun Hong,^{ID b} and Hai-Long Jiang,^{ID *a}

While the rational fabrication of heterojunction photocatalysts with tunable interfaces and precise location control of cocatalysts holds great promise for enhanced photocatalysis, the synergistic integration of these parameters remains a substantial challenge. Herein, a series of metal–organic framework (MOF) composites with compact interfaces and customizable interface coverage are designed by epitaxial growth of ZIF-8 on the surface of MIL-125-NH₂, yielding ZIF-8_m/MIL-125-NH₂ ($m = 21, 35, 65$, representing the coverage percentage of ZIF-8 on the MIL-125-NH₂ surface). These composites are then converted into ZnO/TiO_x heterojunctions through a two-step thermal treatment, termed ZTO- m , for photocatalytic CH₄ oxidation. The results reveal that the interface coverage in ZTO- m plays critical roles in charge separation, where ZTO-65 gives the best activity. With ZTO-65 as a basis, the cocatalysts, Au clusters and CoO_x species, are respectively positioned onto TiO_x and ZnO. The targeted positioning of cocatalysts not only improves charge separation but also facilitates O₂ activation. As a result, the resulting Au-Co-ZTO demonstrates excellent activity toward liquid oxygenate production, achieving 1723.5 $\mu\text{mol g}^{-1} \text{h}^{-1}$ with a selectivity of 99%, in photocatalytic CH₄ oxidation.

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Introduction

Matter transformation, the fundamental process underlying life activities, is an inherently energy-consuming process due to the thermodynamic limitation. The realization of efficient matter transformation under mild conditions represents a long-standing objective in scientific research. Photocatalysis, which harnesses light energy to drive chemical reactions, offers a promising way to overcome thermodynamic barriers and facilitate energy-intensive reactions under mild conditions.^{1–3} To improve the photocatalytic performance, various strategies have been developed,^{4–11} among which constructing heterojunctions and incorporating cocatalysts have emerged as very promising approaches.^{8–11}

In heterojunction systems, interface interaction plays crucial roles in determining charge transfer efficiency.¹² However, challenges such as lattice mismatch and chemical incompatibility often prevent the formation of tightly bonded

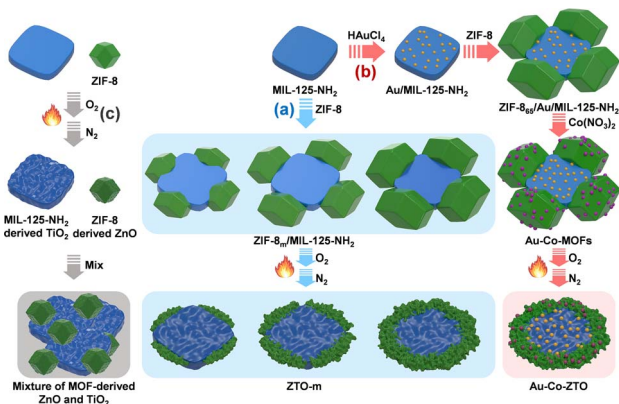
heterojunction interfaces between different components.¹³ As a result, heterojunction photocatalysts are typically constructed by simple physical assembly, which tends to result in poor interface contact and inefficient charge transfer. Achieving a tight, well-structured interface and precisely regulated interface area/length between two semiconductor photocatalysts remains a significant challenge. In addition, cocatalysts are widely recognized for enhancing charge separation and substrate activation in photocatalysis, while their random distribution often diminishes these advantages.¹⁴ This limitation is particularly pronounced in heterojunction photocatalysts with complex components, where precise location control of cocatalysts becomes increasingly challenging. Therefore, the development of heterojunction systems with closely contacted interfaces and precisely positioned cocatalysts could significantly improve the photocatalytic efficiency. Unfortunately, such design has rarely been reported so far, and it remains a grand challenge to achieve using conventional methods.^{15,16}

Metal–organic frameworks (MOFs), distinguished by their extraordinary structural versatility, have attracted widespread attention across diverse application fields.^{17–23} MOFs exhibit high structural compatibility and customizable skeletons, enabling the fabrication of composites with controllable components, sizes, and interfaces.^{24–26} Moreover, MOFs can

^aHefei National Research Center for Physical Sciences at the Microscale, Department of Chemistry, University of Science and Technology of China, Hefei, Anhui 230026, P. R. China. E-mail: jianglab@ustc.edu.cn; Web: <http://mof.ustc.edu.cn/>

^bState Key Laboratory of Structural Chemistry, Fujian Institute of Research on the Structure of Matter, Chinese Academy of Sciences, Fuzhou, Fujian 350002, P. R. China. E-mail: chengqh@fjirsm.ac.cn

[†] These authors contributed equally to this work.



Scheme 1 Illustration showing the stepwise synthesis route to the fabrication of (a) ZTO-*m*, (b) Au-Co-ZTO, and (c) MOF-derived ZnO + TiO₂ mixture photocatalysts for CH₄ oxidation.

serve as versatile precursors for creating different derivatives tailored to diverse applications.^{27–30} More importantly, these MOF-derived materials often inherit the original morphology and interface characteristics of their parent MOFs.^{31–33} By leveraging the inherent tunability of MOFs, it might be possible to fabricate MOF-based composites with tunable interfaces and precise cocatalyst placement as precursors, followed by thermal treatment to afford heterojunction photocatalysts with inherited structural features for improved photocatalysis.

In this work, binary MOF composites with controlled interface coverage were obtained by controlling the epitaxial growth of ZIF-8 on the surface of MIL-125-NH₂, affording ZIF-8_{*m*}/MIL-125-NH₂ (*m* = 21, 35, 65, representing the coverage percentage of ZIF-8 on the MIL-125-NH₂ surface). Subsequently, ZTO-*m* composites featuring varying heterojunction interface coverages between zinc oxide (ZnO) and titanium oxide (TiO_x) were synthesized *via* a two-step thermal treatment of ZIF-8_{*m*}/MIL-125-NH₂ (Scheme 1a). The heterojunction interface area (with *m* as the descriptor; see SI, Discussion S1) is found to significantly influence charge separation, and as a result, ZTO-65 affords optimal performance toward photocatalytic CH₄ oxidation. Further, Au clusters and CoO_x species, serving as cocatalysts, were loaded onto ZIF-8 and MIL-125-NH₂, respectively, to form the Au-Co-MOF composite with a ZIF-8 coverage of 65%, followed by thermal treatment to yield Au-Co-ZTO (Scheme 1b). Results demonstrate that the cocatalyst distribution not only suppresses the recombination of electrons and holes but also plays a crucial role in O₂ activation. Remarkably, the optimized Au-Co-ZTO, featuring Au clusters on TiO_x and CoO_x on ZnO, achieves a production rate of 1723.5 μmol g^{−1} h^{−1} with 99% selectivity toward liquid oxygenate production in photocatalytic CH₄ oxidation.

Results and discussion

Synthesis and characterization

Initially, MIL-125-NH₂ was prepared with titanium isopropoxide and aminoterephthalate *via* a hydrothermal method at 150 °C. Subsequently, ZIF-8 was grown on the surface of MIL-125-NH₂

particles *via* an epitaxial growth strategy.³⁴ By adjusting the amount of ZIF-8 precursor, the ZIF-8_{*m*}/MIL-125-NH₂ composites (*m* indicates the coverage percentage of ZIF-8 on the surface of MIL-125-NH₂ particles, *m* = 21, 35, 65) were obtained. Powder X-ray diffraction (XRD) shows that the intensity ratio of diffraction peaks for ZIF-8 and MIL-125-NH₂ reasonably increases with higher ZIF-8 loading (Fig. S1). Scanning electron microscopy (SEM) observation further confirms the different degrees of ZIF-8 coverage on the MIL-125-NH₂ surface (Fig. 1a and S2). The coverage percentage of ZIF-8 on the MIL-125-NH₂ surface is controlled from 21% to 35%, and up to 65% (Fig. S3). The limitation in coverage is attributed to the inherent growth restrictions of ZIF-8 on the periphery of MIL-125-NH₂.³⁴

The above ZIF-8₆₅/MIL-125-NH₂ composites were transformed into ZTO-65 *via* pyrolysis in O₂ at 425 °C, followed by N₂ treatment at 600 °C (Scheme 1a). The first pyrolysis under an oxygen atmosphere produced metal oxides. During this process, the original morphology of MIL-125-NH₂ is largely inherited in the resulting titanium oxide, while the external ZIF-8 is transformed into porous ZnO with a stacked structure, as previously reported.³⁵ Powder XRD profiles reveal that, while ZIF-8 and MIL-125-NH₂ are respectively converted to ZnO and TiO₂ upon thermal treatment, for ZIF-8₆₅/MIL-125-NH₂, only ZnO diffraction peaks can be observed, with the absence of TiO₂ signals (Fig. S4). This absence of crystalline TiO₂ might be attributed to the influence of Zn during pyrolysis, which inhibits the crystallization of TiO₂, leading to the formation of amorphous TiO_x.³⁶ To enhance the interface interaction, higher pyrolysis temperatures were investigated.³⁷ While temperatures above 500 °C promote the formation of tight ZnO/TiO_x interfaces, too high temperature leads to structural collapse (Fig. S4 and S5). As a result of optimization, the ZIF-8₆₅/MIL-125-NH₂ composite was first treated in an oxygen atmosphere at 425 °C to produce metal oxide products, followed by N₂ treatment at 600 °C. The inert N₂ atmosphere would facilitate compact interfacial contact and avoid structural collapse.³⁸ Such a two-step process yields ZTO-65, in which not only a well-structured

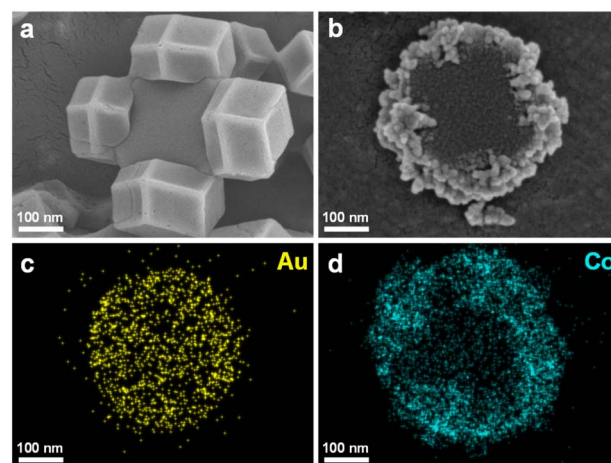


Fig. 1 SEM images of (a) ZIF-8₆₅/MIL-125-NH₂, and (b) Au-Co-ZTO. (c) and (d) EDS mapping of Au and Co elements for Au-Co-ZTO.

heterojunction is maintained but also a new Zn_2TiO_4 phase is created (Fig. S6 and S7).

The emergence of this new phase suggests a compact interface within the heterojunction, facilitating charge separation.³⁹ As verification, photocurrent and electrochemical impedance spectroscopy (EIS) results confirm that **ZTO-65** prepared *via* the two-step thermal treatment possesses superior charge separation compared to that produced by direct pyrolysis (Fig. S8). The results indicate that the two-step process not only largely preserves the morphology of the parent MOFs but also promotes the formation of a compact interface, improving charge separation. Additionally, **ZTO-21** and **ZTO-35**, with different coverage levels, were also prepared as controls under identical transformation conditions (Fig. S9 and S10).

Moreover, two cocatalysts, Au clusters and CoO_x , are introduced into the **ZTO** heterojunction system. The spatial arrangement of these cocatalysts is precisely controlled by varying the introduction sequence of cocatalyst precursors (Scheme 1b). Typically, Au clusters were first photodeposited on the surface of MIL-125-NH₂, followed by the epitaxial growth of ZIF-8, during which the coverage percentage of ZIF-8 remained at the optimized value of 65%. Subsequently, Zn^{2+} within ZIF-8 was exchanged with Co^{2+} through ion exchange, resulting in a Au-Co-MOF with Au and Co localized on MIL-125-NH₂ and ZIF-8, respectively. High-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) images and the corresponding energy dispersive X-ray spectroscopy (EDS) mapping show that Au clusters are located on MIL-125-NH₂ and the Co element is on ZIF-8 (Fig. S11). Control experiments confirmed that the Co element cannot be incorporated into MIL-125-NH₂ by the ion exchange process, ensuring its incorporation into ZIF-8 only (Table S1).

The obtained Au-Co-MOF composite with well-defined Au/Co locations is then converted into the corresponding oxide, namely **Au-Co-ZTO**, with the above two-step thermal treatment procedure as that for **ZTO-65**. Electron microscopy observation and powder XRD results indicate that the crystallinity and morphology of **Au-Co-ZTO** are similar to those of **ZTO-65** (Fig. 1b, S12 and S13). X-ray photoelectron spectroscopy (XPS)

reveals that Au is in its metallic state, while Co exhibits a mixed oxidation state of +2 and +3, identified as CoO_x (Fig. S14).⁴⁰ Furthermore, the Au clusters are hardly observed, indicating their tiny sizes, and primarily located on the central TiO_x based on EDS mapping results, whereas CoO_x species are predominantly distributed on the peripheral ZnO (Fig. S15, 1c and d). For comparison, **ZTO-65** with randomly distributed cocatalysts, referred to as **Au+Co+ZTO**, was also obtained by random impregnation of Au and Co species into **ZTO-65** (Fig. S16 and S17).

Photocatalytic performance for methane oxidation

Encouraged by the above engineered interface coverage and precise cocatalyst placement, photocatalytic methane oxidation, which enables the conversion of CH_4 into value-added oxygenates, has been conducted to investigate the significance of such structural design.^{41–45} The experiment was investigated using O_2 as the oxidizing agent and H_2O as the solvent. While the single-component ZIF-8-derived ZnO catalyst demonstrates the ability to selectively convert CH_4 into liquid oxygenates, including CH_3OH , CH_3OOH , HCHO , and HCOOH , it gives very low activity (Table 1, entry 1; Fig. S18 and S19). In contrast, MIL-125-NH₂-derived TiO_2 shows higher activity than the above ZIF-8-derived ZnO; however, it readily generates over-oxidized CH_4 to CO_2 , resulting in low selectivity toward liquid oxygenate production (Table 1, entry 2).

Based on the high activity of MIL-125-NH₂-derived TiO_2 and high selectivity of ZIF-8-derived ZnO, it is assumed that **ZTO-*m*** composites could combine their respective advantages. Compared to their single-component counterparts, **ZTO-21** presents slightly increased activity for liquid oxygenate production with high selectivity, highlighting the importance of the heterojunction in promoting CH_4 oxidation (Table 1, entry 3). Furthermore, as the coverage increases, the photocatalytic activity improves in the order of **ZTO-65** > **ZTO-35** > **ZTO-21** (Table 1, entry 3–5). Among these composites, **ZTO-65** shows the best performance, achieving an activity of $628.7 \mu\text{mol g}^{-1} \text{h}^{-1}$, which is 1.9 times higher than that of ZIF-8-derived ZnO and 1.4 times higher than that of MIL-125-NH₂-derived TiO_2 . Notably,

Table 1 Performance of photocatalytic methane oxidation over different catalysts derived from MOFs^a

Entry	Catalyst	Yield ($\mu\text{mol g}^{-1} \text{h}^{-1}$)					Liquid products	Liquid product selectivity
		CO_2	CH_3OH	CH_3OOH	HCHO	HCOOH		
1	ZIF-8-derived ZnO	0	51.2	168.9	96.4	13.2	329.7	1.00
2	MIL-125-NH ₂ -derived TiO_2	152.3	95.4	15.6	325.0	14.8	450.8	0.75
3	ZTO-21	9.0	363.8	25.7	82.7	3.9	476.1	0.98
4	ZTO-35	3.7	128.6	176.7	248.0	7.8	561.1	0.99
5	ZTO-65	0	58.9	391.5	163.6	14.7	628.7	1.00
6	Mixture of MOF-derived ZnO and TiO_2	40.0	26.0	114.3	167.9	11.7	319.9	0.89
7	Au-Co-ZTO	18.9	145.5	1195.2	360.2	22.6	1723.5	0.99
8	Au+Co+ZTO	41.3	421.4	331.0	295.2	35.9	1083.5	0.96

^a Reaction conditions: catalyst (5 mg), O_2 (1 bar), CH_4 (20 bar), H_2O (10 mL), 25 °C, 300 W Xe lamp (320–800 nm).



ZTO-65 not only exhibits higher activity, but also gives 100% selectivity toward liquid oxygenate production. As a control, the physical mixture of the MOF-derived ZnO and TiO₂ shows much lower activity and selectivity than the heterojunction composites (Table 1, entry 6; Fig. S20). These results unambiguously underscore the critical role of interface coverage for selective CH₄ oxidation.

The above results reveal that the interface coverage within heterojunction photocatalysts can be engineered using highly designable MOF composites as precursors (see SI, Discussion S1), which significantly influences the photocatalytic performance in CH₄ oxidation. To further improve CH₄ oxidation activity, Au and CoO_x cocatalysts have been further introduced. First, the photocatalytic performance of ZTO-65 with selectively loaded Au clusters on the TiO_x component has been evaluated. Along with increased loading of Au clusters, the photocatalytic activity follows a volcano-type trend (Fig. S21a). At the optimal Au loading of 0.58 wt%, the CoO_x cocatalyst was further introduced into the ZnO component, which indicates a similar activity trend as observed with Au loading change (Fig. S21b). When the Au and Co loadings reach 0.58 wt% and 0.39 wt%, respectively, the resulting Au-Co-ZTO exhibits the highest activity, achieving a liquid oxygenate production rate of 1723.5 $\mu\text{mol g}^{-1} \text{h}^{-1}$ with 99% selectivity (Table 1, entry 7). Compared with previous reports,^{40,46–48} the excellent performance of the heterojunction system with oxygen as the oxidant is significant

(Table S2). In contrast, the **Au+Co+ZTO** catalyst, where Au and CoO_x cocatalysts are randomly distributed, showcases significantly lower activity than **Au-Co-ZTO**, highlighting the significance of precisely loaded and placed cocatalysts for enhanced CH₄ oxidation (Table 1, entry 8).

Furthermore, an isotope labeling experiment has been conducted to trace the source of liquid oxygenate production (Fig. 2a). When pure ¹²CH₄ is used as the feedstock, two distinct single peaks are observed at 3.18 ppm and 3.68 ppm in the ¹H-nuclear magnetic resonance (¹H-NMR) spectrum, corresponding to ¹²CH₃OH and ¹²CH₃OOH. In contrast, when a mixed gas of ¹³CH₄:¹²CH₄ in a 1:2 ratio is employed to replace pure ¹²CH₄, two split peaks appear due to the coupling splitting caused by the presence of ¹³C.⁴⁹ Additionally, the ratio of ¹³CH₃OH/¹²CH₃OH is 1:2, consistent with the ratio of feeding gases. These results clearly confirm that the liquid oxygenate production originates from CH₄, rather than the decomposition of residual carbon. The stability of **Au-Co-ZTO** during photocatalysis has also been examined. After 5 consecutive photocatalytic cycles, no significant changes are observed in both activity and selectivity (Fig. 2b). Characterization confirms that the crystallinity and integrity are well preserved, suggesting the stability of the **Au-Co-ZTO** photocatalyst (Fig. S22–S25 and Table S3).

Mechanism investigations

To unveil the mechanisms behind the enhanced photocatalytic performance of **Au-Co-ZTO**, charge transfer and CH₄ activation processes have been investigated. First, *in situ* XPS was employed to elucidate the charge transfer pathway in **ZTO-65**. Upon light irradiation, the Ti 2p binding energy shifts by −0.2 eV (Fig. 2c), whereas the Zn 2p binding energy increases by +0.2 eV (Fig. 2d). The results support that TiO_x and ZnO form a heterojunction system, where electron transfer from ZnO to TiO_x occurs, inferring that holes migrate from high oxidation potential TiO_x to ZnO. Considering the relatively moderate oxidation capability of ZnO, the accumulation of holes in ZIF-8-derived ZnO mitigates the over-oxidation of CH₄ to CO₂ (Table 1, entry 1). The XPS analysis reveals that **ZTO-65** forms compact interfaces, which are more favorable for charge transfer compared with MOF-derived ZnO and TiO₂ (Fig. S26).

Additionally, the large interface areas within this heterojunction structure facilitate charge separation, contributing to the enhanced photocatalysis. All **ZTO-*m*** samples display similar light absorption properties, confirming that the differences in photocatalytic activity are not due to variations in light harvesting (Fig. S27). Steady-state photoluminescence (PL) spectroscopy reveals that **ZTO-65** exhibits the weakest fluorescence intensity (Fig. S28), suggesting the most efficient electron–hole separation among **ZTO-*m***. Similarly, photocurrent measurements and EIS also suggest that **ZTO-65** possesses the best charge separation efficiency and the lowest interface charge transfer resistance (Fig. S29 and S30). Furthermore, time-resolved transient photoluminescence reveals that **ZTO-65** has the longest electron lifetime (0.35 ns), followed by **ZTO-35** (0.30 ns) and **ZTO-21** (0.29 ns) (Fig. S31 and S32). The above results collectively demonstrate that the superior photocatalytic

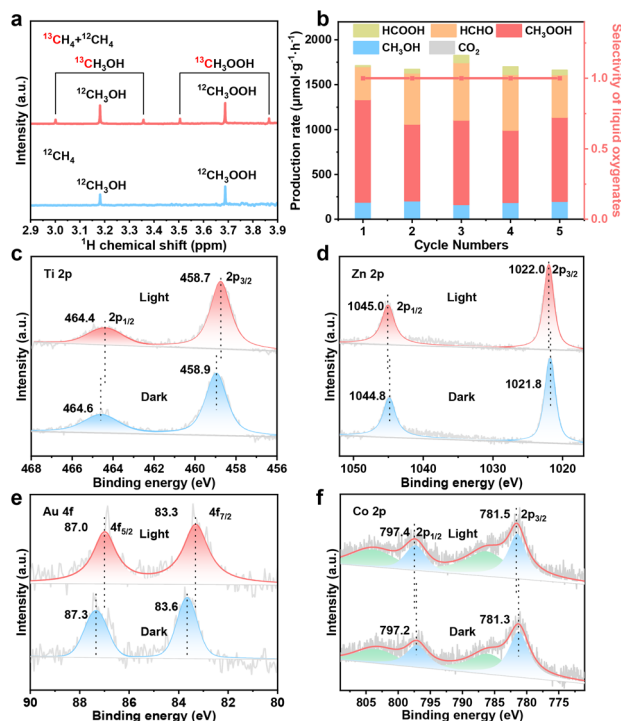


Fig. 2 (a) ¹H NMR spectra of the CH₄ oxidation reaction carried out with a ¹³CH₄ and ¹²CH₄ mixture (33.3% ¹³CH₄ and 66.7% ¹²CH₄) or ¹²CH₄. (b) Recycling performance of Au-Co-ZTO. *In situ* XPS spectra for (c) Ti 2p, (d) Zn 2p of ZTO-65 and (e) Au 4f of the sample obtained by thermal treatment of Au/MIL-125-NH₂, as well as (f) Co 2p of Au-Co-ZTO in the dark and under light irradiation.



performance in **ZTO-65** arises from its large interface coverage in the heterojunction structure, which greatly enhances separation of electrons and holes, thereby benefiting the photocatalysis.

Moreover, the importance of precise cocatalyst placement in promoting charge separation has been investigated. Due to the overlap between Au 4f and Zn 3p XPS signals,⁴⁵ the sample directly from Au/MIL-125-NH₂ (similar thermal treatment to that for preparing **Au-Co-ZTO**) was employed instead to study the electronic state changes of Au under light irradiation. *In situ* XPS spectra reveal that, under light irradiation, the Au and Co peaks shift by -0.3 eV and $+0.2$ eV, respectively, indicating that Au clusters and CoO_x function as reduction and oxidation cocatalysts, respectively (Fig. 2e and f). Therefore, the electron reduction reaction occurs on Au, while the hole oxidation reaction takes place on CoO_x (Fig. S33).⁵⁰ In the heterojunction system, the spatial location of Au and CoO_x would greatly impact charge separation efficiency. All steady-state PL, photocurrent, and EIS measurements confirm that **Au-Co-ZTO** exhibits superior charge separation efficiency compared to **Au+Co+ZTO** with Au and CoO_x randomly distributed (Fig. S34). Time-resolved transient PL spectroscopy further indicates that **Au-Co-ZTO** has a longer fluorescence lifetime than **ZTO-m** without cocatalysts and **Au+Co+ZTO** (Fig. S35 and S36). These observations support that the precise cocatalyst placement is critical for promoting charge separation in heterojunction photocatalysts.

In addition to charge separation, the CH₄ activation process also plays a key role in photocatalytic CH₄ oxidation. The oxygen source for liquid oxygenate production was determined to elucidate the reaction pathway. With **Au-Co-ZTO** as the catalyst, mass spectrometry (MS) analysis of the liquid oxygenate

production gives the CH₃OH signal only, as CH₃OOH is unstable under the testing conditions.⁵¹ The signals of CH₃¹⁸OH and CH₃¹⁶OH can be observed when ¹⁸O₂ and H₂¹⁶O are used (Fig. 3a), suggesting that the oxygen in the products originates from both water and oxygen.

Furthermore, electron paramagnetic resonance (EPR) spectroscopy was employed to identify the reactive oxygen species (ROS) involved in the photocatalytic process. To identify signals originating from oxidation reactions, the catalysts were dispersed in water, and AgNO₃ was added to suppress interference from reduction-related ROS, such as [•]OOH.⁵² No signal is detected in the dark when DMPO is used as the radical trapping agent (Fig. 3b). Upon light irradiation, DMPO-[•]OH signals are observed in the presence of **Au-Co-ZTO**, indicating the production of hydroxyl radicals ([•]OH) derived from H₂O oxidation.⁵² Similarly, the catalysts were dispersed in methanol (as a hole sacrificial agent) to investigate the ROS produced from reduction reactions. In this case, only DMPO-[•]OOH signals are detected, while [•]OH signals are absent (Fig. 3c). The DMPO-[•]OOH signals suggest that O₂ reacts with an electron and H⁺ to form hydroperoxyl radical ([•]OOH) species, which are milder ROS than superoxide anions (O₂^{•-}).⁵³ These EPR results demonstrate the formation of [•]OH and [•]OOH, where the former is produced from water oxidation by holes, while the latter results from the reduction of O₂ by electrons.

To further elucidate the roles of [•]OH and [•]OOH, quenching experiments were conducted (Fig. S37). The addition of salicylic acid, a [•]OH scavenger, significantly suppresses product formation, indicating that CH₄ activation is primarily driven by [•]OH. When Na₂C₂O₄, the hole scavenger, is added, the activity decreases but is not completely suppressed. Introducing H₂O₂ partially restores the activity, suggesting that water oxidation might not be the sole pathway for the [•]OH production. A smaller portion is generated from the decomposition of H₂O₂, which is produced from the further conversion of partial [•]OOH.⁵⁴ Unfortunately, [•]OH from H₂O₂ is hardly detected by EPR due to its low concentration.⁵⁵

According to these results, a potential photocatalytic CH₄ oxidation mechanism is proposed (Fig. 3d). During photocatalysis, photogenerated holes oxidize H₂O to produce [•]OH, while electrons reduce O₂ to form [•]OOH. A fraction of [•]OOH undergoes hydrogenation to yield H₂O₂, which subsequently decomposes to release additional [•]OH.⁵⁶ The generated [•]OH abstracts a hydrogen atom from CH₄, forming methyl radicals ([•]CH₃) and initiating the methane activation (step I).⁵⁷ The [•]CH₃ radicals then react with [•]OOH to produce CH₃OOH, which undergoes further transformation to CH₃OH (step II, III). In addition, CH₃OH also can be formed through the direct reaction between [•]OH and [•]CH₃ (step IV).⁵⁷ Dehydration of CH₃OOH or further oxidation of CH₃OH results in HCHO, which can be subsequently oxidized to HCOOH and eventually CO₂ (step V–VIII).

Based on the reaction mechanism, the production rates of [•]OH and [•]OOH were quantified to unveil the influence of cocatalyst placement on surface reactions. EPR spectra show similar ROS production over **Au+Co+ZTO** and **Au-Co-ZTO**, indicating that the reaction pathway is undisturbed by the

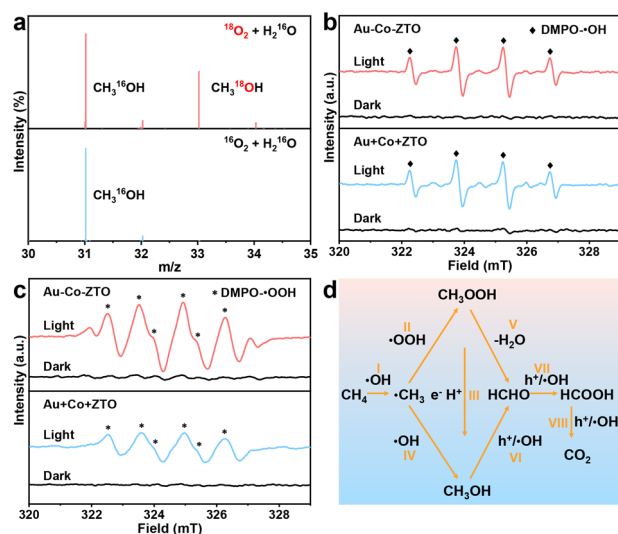
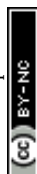


Fig. 3 (a) MS results of the isotope labeling experiments in the presence of ¹⁸O₂ + H₂¹⁶O or ¹⁶O₂ + H₂¹⁶O over **Au-Co-ZTO**. EPR spectra of (b) DMPO-[•]OH and (c) DMPO-[•]OOH for monitoring the generation of [•]OH and [•]OOH active species over **Au-Co-ZTO** and **Au+Co+ZTO**. (d) Schematic illustration of the mechanism of photocatalytic CH₄ conversion.



cocatalyst placement (Fig. 3b and c). However, while the DMPO- $\cdot\text{OH}$ signal intensity is comparable, the DMPO- $\cdot\text{OOH}$ intensity is obviously weaker for **Au+Co+ZTO** than **Au-Co-ZTO**. This suggests that the cocatalyst placement significantly impacts the oxygen reduction to $\cdot\text{OOH}$ but has a minimal impact on H_2O oxidation to $\cdot\text{OH}$, which avoids over-oxidation of liquid oxygenate production by excess $\cdot\text{OH}$. Furthermore, the production rate of different ROS was quantified by probe experiments for the two catalysts. Specifically, coumarin acted as the fluorescence probe for $\cdot\text{OH}$ detection, and the degradation rate of nitrotetrazolium blue chloride (NBT) was adopted to examine $\cdot\text{OOH}$.⁴³ The PL intensity growth rate of 7-hydroxycoumarin originating from the reaction between coumarin and $\cdot\text{OH}$ is nearly equal, supporting that the placement of cocatalysts does not affect the water oxidation to $\cdot\text{OH}$ (Fig. 4a and S38). However, the NBT degradation rate, indicative of $\cdot\text{OOH}$ production, is higher for **Au-Co-ZTO**, consistent with the EPR results (Fig. 4b and S39). These findings demonstrate that the cocatalyst placement not only improves the charge separation, but also accelerates the O_2 activation to $\cdot\text{OOH}$ for boosting photocatalytic activity.

In addition, mechanisms regarding how the cocatalyst placement influences CH_4 activation have been investigated. XPS analysis was employed to disclose the electronic states of Au and Co, which affect the activation of intermediates. The Au 4f_{7/2} signal for **Au+Co+ZTO** appears at 83.5 eV, while it shifts to 83.3 eV for **Au-Co-ZTO**, indicating an increased electronic density of Au in the latter (Fig. 4c). This increased electronic density of Au clusters in **Au-Co-ZTO** enables efficient electron donation to O_2 , thereby facilitating activation of O_2 .⁵⁸ In contrast, no difference is observed in the Co XPS signals between the two catalysts (Fig. 4d). This indicates that the absorption and activation of H_2O on CoO_x are comparable,

consistent with their similar abilities to produce $\cdot\text{OH}$ from water oxidation (Fig. 3b and 4a). These results suggest that the cocatalyst distribution modulates the electronic states of Au, further enhancing O_2 activation for improved CH_4 oxidation.

Conclusions

In summary, heterojunction photocatalysts with engineered surface coverage and precise cocatalyst placement have been fabricated by pre-assembly of binary MOF composites, followed by two-step thermal treatment. ZnO coverage on the TiO_x surface in the heterojunction composites is systematically regulated by varying the interface coverage between their respective MOF precursors, resulting in **ZTO-*m*** ($m = 21, 35, 65$), for photocatalytic CH_4 oxidation. It is observed that **ZTO-65** with the largest interface coverage between the two components significantly improves charge separation and displays the highest activity with 100% selectivity toward liquid oxygenate production, disclosing the critical role of interface coverage in photocatalysis.

On the basis of **ZTO-65** with optimized interface coverage, the cocatalysts, Au clusters and CoO_x , are further deposited on TiO_x and ZnO, respectively. The spatial distribution of cocatalysts is found to affect charge separation, and meanwhile, improve O_2 activation, further optimizing the performance for photocatalytic CH_4 oxidation. Accordingly, the optimized **Au-Co-ZTO** achieves a liquid oxygenate production yield of $1723.5 \mu\text{mol g}^{-1} \text{h}^{-1}$ with an impressive selectivity of 99%. This work not only highlights the great potential of MOFs toward photocatalysis, but also offers deep insights into the development of heterostructured photocatalytic materials.

Author contributions

H.-L. J. and Q. C. supervised the project. W. Z., K. S. and J. X. performed all synthetic and characterization experiments. Z. L. assisted in figures. M. H. provided valuable discussions. W. Z., K. S. and H.-L. J. co-wrote the manuscript. All authors discussed the results and commented on the manuscript.

Conflicts of interest

There are no conflicts to declare.

Data availability

The additional figures, tables, and data supporting this article have been included as part of the SI. Supplementary information is available. See DOI: <https://doi.org/10.1039/d5sc04771d>.

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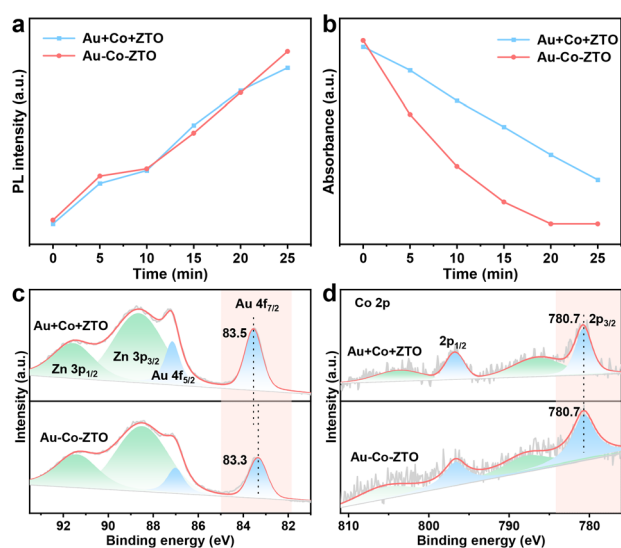


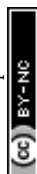
Fig. 4 (a) Time-dependent PL intensity of the produced 7-hydroxycoumarin for $\cdot\text{OH}$ radical detection. (b) Time-dependent absorption intensity of NBT for $\cdot\text{OOH}$ radical detection. XPS spectra for (c) Au 4f and (d) Co 2p of **Au-Co-ZTO** and **Au+Co+ZTO**.



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References

- 1 S. Shoji, X. Peng, A. Yamaguchi, R. Watanabe, C. Fukuhara, Y. Cho, T. Yamamoto, S. Matsumura, M.-W. Yu, S. Ishii, T. Fujita, H. Abe and M. Miyauchi, *Nat. Catal.*, 2020, **3**, 148–153.
- 2 Y. Yuan, L. Zhou, H. Robatjazi, J. L. Bao, J. Zhou, A. Bayles, L. Yuan, M. Lou, M. Lou, S. Khatiwada, E. A. Carter, P. Nordlander and N. J. Halas, *Science*, 2022, **378**, 889–893.
- 3 K. Sun, Y. Huang, F. Sun, Q. Wang, Y. Zhou, J. Wang, Q. Zhang, X. Zheng, F. Fan, Y. Luo, J. Jiang and H.-L. Jiang, *Nat. Chem.*, 2024, **16**, 1638–1646.
- 4 F. Chen, T. Ma, T. Zhang, Y. Zhang and H. Huang, *Adv. Mater.*, 2021, **33**, 2005256.
- 5 L. Zhang, J. Liu and Y.-Q. Lan, *Acc. Chem. Res.*, 2024, **57**, 870–883.
- 6 S. Bhattacharjee, S. Linley and E. Reisner, *Nat. Rev. Chem.*, 2024, **8**, 87–105.
- 7 R. S. H. Khoo, C. Fiankor, S. Yang, W. Hu, C. Yang, J. Lu, M. D. Morton, X. Zhang, Y. Liu, J. Huang and J. Zhang, *J. Am. Chem. Soc.*, 2023, **145**, 24052–24060.
- 8 B. Zhu, J. Sun, Y. Zhao, L. Zhang and J. Yu, *Adv. Mater.*, 2024, **36**, 2310600.
- 9 A. Dhakshinamoorthy, Z. Li, S. Yang and H. Garcia, *Chem. Soc. Rev.*, 2024, **53**, 3002–3035.
- 10 J. Yang, D. Wang, H. Han and C. Li, *Acc. Chem. Res.*, 2013, **46**, 1900–1909.
- 11 Q. Zhang, C. Yang, Y. Chen, Y. Yan, M. Kan, H. Wang, X. Lv, Q. Han and G. Zheng, *Angew. Chem., Int. Ed.*, 2025, **64**, e202419282.
- 12 A. V. Akimov, A. J. Neukirch and O. V. Prezhdo, *Chem. Rev.*, 2013, **113**, 4496–4565.
- 13 P. Wang, C. Jia, Y. Huang and X. Duan, *Matter*, 2021, **4**, 552–581.
- 14 B. Qiu, M. Du, Y. Ma, Q. Zhu, M. Xing and J. Zhang, *Energy Environ. Sci.*, 2021, **14**, 5260–5288.
- 15 Y. Qi, J. Zhang, Y. Kong, Y. Zhao, S. Chen, D. Li, W. Liu, Y. Chen, T. Xie, J. Cui, C. Li, K. Domen and F. Zhang, *Nat. Commun.*, 2022, **13**, 484.
- 16 M.-Y. Qi, Y.-H. Li, F. Zhang, Z.-R. Tang, Y. Xiong and Y.-J. Xu, *ACS Catal.*, 2020, **10**, 3194–3202.
- 17 H. Furukawa, K. E. Cordova, M. O’Keeffe and O. M. Yaghi, *Science*, 2013, **341**, 1230444.
- 18 H.-C. J. Zhou and S. Kitagawa, *Chem. Soc. Rev.*, 2014, **43**, 5415–5418.
- 19 Y. Shen, T. Pan, L. Wang, Z. Ren, W. Zhang and F. Huo, *Adv. Mater.*, 2021, **33**, 2007442.
- 20 X. Zhao, Y. Wang, D.-S. Li, X. Bu and P. Feng, *Adv. Mater.*, 2018, **30**, 1705189.
- 21 Q.-J. Wu, J. Liang, Y.-B. Huang and R. Cao, *Acc. Chem. Res.*, 2022, **55**, 2978–2997.
- 22 T. Zhang and W. Lin, *Chem. Soc. Rev.*, 2014, **43**, 5982–5993.
- 23 G. Li, S. Zhao, Y. Zhang and Z. Tang, *Adv. Mater.*, 2018, **30**, 1800702.
- 24 J. Liu, T. A. Goetjen, Q. Wang, J. G. Knapp, M. C. Wasson, Y. Yang, Z. H. Syed, M. Delferro, J. M. Notestein, O. K. Farha and J. T. Hupp, *Chem. Soc. Rev.*, 2022, **51**, 1045–1097.
- 25 L. Jiao, J. Wang and H.-L. Jiang, *Acc. Mater. Res.*, 2021, **2**, 327–339.
- 26 R.-B. Lin, Z. Zhang and B. Chen, *Acc. Chem. Res.*, 2021, **54**, 3362–3376.
- 27 Y.-Z. Chen, R. Zhang, L. Jiao and H.-L. Jiang, *Coord. Chem. Rev.*, 2018, **362**, 1–23.
- 28 K. J. Lee, J. H. Lee, S. Jeoung and H. R. Moon, *Acc. Chem. Res.*, 2017, **50**, 2684–2692.
- 29 Y.-T. Zheng, S. Li, N.-Y. Huang, X. Li and Q. Xu, *Coord. Chem. Rev.*, 2024, **510**, 215858.
- 30 T. De Villenoisy, X. Zheng, V. Wong, S. S. Mofarah, H. Arandiyani, Y. Yamauchi, P. Koshy and C. C. Sorrell, *Adv. Mater.*, 2023, **35**, 2210166.
- 31 L. Chai, R. Li, Y. Sun, K. Zhou and J. Pan, *Adv. Mater.*, 2025, **37**, 2413658.
- 32 C. Wang, Y. Yao, J. Li and Y. Yamauchi, *Acc. Mater. Res.*, 2022, **3**, 426–438.
- 33 G. Cai, P. Yan, L. Zhang, H.-C. Zhou and H.-L. Jiang, *Chem. Rev.*, 2021, **121**, 12278–12326.
- 34 C. Liu, T. Bao, L. Yuan, C. Zhang, J. Wang, J. Wan and C. Yu, *Adv. Funct. Mater.*, 2022, **32**, 2111404.
- 35 S. Park, D. Oh, J. Ahn, J. K. Kim, D.-H. Kim, S. Kim, C. Park, W. Jung and I.-D. Kim, *Adv. Mater.*, 2022, **34**, 2201109.
- 36 C. Chuaicham, S. Karthikeyan, J. T. Song, T. Ishihara, B. Ohtani and K. Sasaki, *ACS Appl. Mater. Interfaces*, 2020, **12**, 9169–9180.
- 37 K. S. Ranjith and T. Uyar, *CrystEngComm*, 2018, **20**, 5801–5813.
- 38 F. Cao, Y. Zhang, H. Wang, K. Khan, A. K. Tareen, W. Qian, H. Zhang and H. Ågren, *Adv. Mater.*, 2022, **34**, 2107554.
- 39 Y.-C. Liang, C.-Y. Hu and Y.-C. Liang, *CrystEngComm*, 2012, **14**, 5579–5584.
- 40 H. Song, X. Meng, S. Wang, W. Zhou, S. Song, T. Kako and J. Ye, *ACS Catal.*, 2020, **10**, 14318–14326.
- 41 B. An, Z. Li, Z. Wang, X. Zeng, X. Han, Y. Cheng, A. M. Sheveleva, Z. Zhang, F. Tuna, E. J. L. McInnes, M. D. Frogley, A. J. Ramirez-Cuesta, L. S. Natrajan, C. Wang, W. Lin, S. Yang and M. Schröder, *Nat. Mater.*, 2022, **21**, 932–938.
- 42 C. Feng, S. Zuo, M. Hu, Y. Ren, L. Xia, J. Luo, C. Zou, S. Wang, Y. Zhu, M. Rueping, Y. Han and H. Zhang, *Nat. Commun.*, 2024, **15**, 9088.
- 43 L. Luo, Z. Gong, Y. Xu, J. Ma, H. Liu, J. Xing and J. Tang, *J. Am. Chem. Soc.*, 2022, **144**, 740–750.
- 44 S. Song, H. Song, L. Li, S. Wang, W. Chu, K. Peng, X. Meng, Q. Wang, B. Deng, Q. Liu, Z. Wang, Y. Weng, H. Hu, H. Lin, T. Kako and J. Ye, *Nat. Catal.*, 2021, **4**, 1032–1042.
- 45 P. Wang, R. Shi, Y. Zhao, Z. Li, J. Zhao, J. Zhao, G. I. N. Waterhouse, L.-Z. Wu and T. Zhang, *Angew. Chem., Int. Ed.*, 2023, **62**, e202304301.



- 46 Y. Jiang, W. Zhao, S. Li, S. Wang, Y. Fan, F. Wang, X. Qiu, Y. Zhu, Y. Zhang, C. Long and Z. Tang, *J. Am. Chem. Soc.*, 2022, **144**, 15977–15987.
- 47 X. Sun, X. Chen, C. Fu, Q. Yu, X.-S. Zheng, F. Fang, Y. Liu, J. Zhu, W. Zhang and W. Huang, *Nat. Commun.*, 2022, **13**, 6677.
- 48 Y. Cao, W. Yu, Y. Li, J. Meng, K. Zheng, C. Huang, X. Yang, Y. Yang, F. Dong and Y. Zhou, *Adv. Energy Mater.*, 2025, **15**, 2404871.
- 49 V. Fernandes de Almeida, S. Navalón, A. Dhakshinamoorthy and H. Garcia, *Angew. Chem., Int. Ed.*, 2025, **64**, e202424537.
- 50 T. Zhou, J. Wang, S. Chen, J. Bai, J. Li, Y. Zhang, L. Li, L. Xia, M. Rahim, Q. Xu and B. Zhou, *Appl. Catal. B*, 2020, **267**, 118599.
- 51 Y. Jiang, S. Li, S. Wang, Y. Zhang, C. Long, J. Xie, X. Fan, W. Zhao, P. Xu, Y. Fan, C. Cui and Z. Tang, *J. Am. Chem. Soc.*, 2023, **145**, 2698–2707.
- 52 C. Xu, Y. Pan, G. Wan, H. Liu, L. Wang, H. Zhou, S.-H. Yu and H.-L. Jiang, *J. Am. Chem. Soc.*, 2019, **141**, 19110–19117.
- 53 J. Wang and S. Wang, *Chem. Eng. J.*, 2020, **401**, 126158.
- 54 Q. Zhang, K. Gu, C. Dong, C. Xue, H. Che, K. Zhang and Y. Ao, *Angew. Chem., Int. Ed.*, 2025, **64**, e202417591.
- 55 J.-N. Chang, Q. Li, J.-W. Shi, M. Zhang, L. Zhang, S. Li, Y. Chen, S.-L. Li and Y.-Q. Lan, *Angew. Chem., Int. Ed.*, 2023, **62**, e202218868.
- 56 Y. Nosaka and A. Y. Nosaka, *Chem. Rev.*, 2017, **117**, 11302–11336.
- 57 J. Sui, M.-L. Gao, B. Qian, C. Liu, Y. Pan, Z. Meng, D. Yuan and H.-L. Jiang, *Sci. Bull.*, 2023, **68**, 1886–1893.
- 58 W. Yang, F. Qi, W. An, H. Yu, S. Liu, P. Ma, R. Chen, S. Liu, L.-L. Lou and K. Yu, *ACS Catal.*, 2024, **14**, 5936–5948.

