

Cite this: *Chem. Sci.*, 2021, 12, 2108

All publication charges for this article have been paid for by the Royal Society of Chemistry

# Fast oxygen ion migration in Cu–In–oxide bulk and its utilization for effective CO<sub>2</sub> conversion at lower temperature†

Jun-Ichiro Makiura,<sup>a</sup> Takuma Higo,<sup>a</sup> \*<sup>a</sup> Yutaro Kurosawa,<sup>a</sup> Kota Murakami,<sup>a</sup> Shuhei Ogo,<sup>a</sup> <sup>a</sup> Hideaki Tsuneki,<sup>a</sup> Yasushi Hashimoto,<sup>a</sup> Yasushi Sato<sup>b</sup> and Yasushi Sekine \*<sup>a</sup>

Efficient activation of CO<sub>2</sub> at low temperature was achieved by reverse water–gas shift *via* chemical looping (RWGS-CL) by virtue of fast oxygen ion migration in a Cu–In structured oxide, even at lower temperatures. Results show that a novel Cu–In<sub>2</sub>O<sub>3</sub> structured oxide can show a remarkably higher CO<sub>2</sub> splitting rate than ever reported. Various analyses revealed that RWGS-CL on Cu–In<sub>2</sub>O<sub>3</sub> is derived from redox between Cu–In<sub>2</sub>O<sub>3</sub> and Cu–In alloy. Key factors for high CO<sub>2</sub> splitting rate were fast migration of oxide ions in the alloy and the preferential oxidation of the interface of alloy–In<sub>2</sub>O<sub>3</sub> in the bulk of the particles. The findings reported herein can open up new avenues to achieve effective CO<sub>2</sub> conversion at lower temperatures.

Received 26th September 2020  
Accepted 10th December 2020

DOI: 10.1039/d0sc05340f

rsc.li/chemical-science

## Introduction

Anthropogenic emissions of greenhouse gases are regarded as a cause of global warming, which is expected to lead to severe future climate change. Particularly, carbon dioxide (CO<sub>2</sub>) derived from fossil fuels and industrial processes has long presented strong effects that continue to raise global mean temperatures.<sup>1,2</sup> Therefore, development of CO<sub>2</sub> capture and utilization (CCU) technologies<sup>2–9</sup> represents an urgent task for reducing CO<sub>2</sub> emissions into the atmosphere and for establishing a sustainable carbon cycle. Synthesis of fuels using electricity generation from renewable resources and using CO<sub>2</sub>, most notably sun-to-fuel (STF)<sup>5</sup> and power-to-liquid<sup>6</sup> processes, has been proposed as a means of developing CCU technologies. In fact, CO<sub>2</sub>-based fuels are anticipated as a technology that can increasingly incorporate renewable energy into the mobility sector as an “e-fuel”.<sup>8,9</sup> For these fuel production technologies, the potential process route is a Fischer–Tropsch (FT) process using syngas, which includes hydrogen and carbon monoxide (CO) converted from captured CO<sub>2</sub>.<sup>2–4</sup> As a method of efficient conversion of CO<sub>2</sub> to CO through this process, reverse water–gas shift (RWGS) is a promising reaction that uses renewable H<sub>2</sub>.<sup>10,11</sup> Mallapragada *et al.* proposed that the STF route consisting of RWGS followed by the FT reaction has higher efficiency than other routes with direct CO<sub>2</sub> conversion by photosynthetic

bacteria and biomass conversion.<sup>5</sup> The RWGS is an equilibrium-limited reaction (eqn (1)) with an endothermic nature.



This reaction requires high reaction temperatures and separation of gas products to gain high conversion and a suitable H<sub>2</sub>/CO ratio in syngas. Furthermore, conventional catalytic RWGS processes include side reactions, as presented in eqn (2) and (3).



The CH<sub>4</sub> produced *via* these side reactions makes gas separation more complex. Additionally, it loses energy because of the exothermic reaction. S.-C. Yang *et al.* reported a Cu/CeO<sub>x</sub> catalyst<sup>12</sup> which demonstrates 100% CO selectivity at 573 K. However, at low temperatures, CO<sub>2</sub> conversion is low because RWGS is an equilibrium-limited endothermic reaction, which results in a much higher H<sub>2</sub>/CO ratio of the obtained syngas than the suitable ratio (H<sub>2</sub>/CO = 2) for fuel synthesis by the FT process. One solution for these shortcomings is using chemical-looping reverse water–gas shift (RWGS-CL).<sup>13,14</sup> The overall concept of the solar-driven fuel (chemical) synthesis process with RWGS-CL is presented in Fig. 1. Through this process, CO<sub>2</sub> is converted to CO in two separate steps: reduction and re-oxidation of a metal oxide as an oxygen storage material (OSM). First, the OSM is reduced by H<sub>2</sub> (reaction (4)). Subsequently, the reduced OSM is re-oxidized by CO<sub>2</sub> to generate pure CO (reaction (5)).

<sup>a</sup>Department of Applied Chemistry, Waseda University, Tokyo 169-8555, Japan. E-mail: t-higo@aoni.waseda.jp; ysekine@waseda.jp

<sup>b</sup>ENEOS, 1-1-2 Otemachi, Chiyoda, Tokyo 100 8162, Japan

† Electronic supplementary information (ESI) available: Experimental procedure, supporting data *etc.* See DOI: 10.1039/d0sc05340f



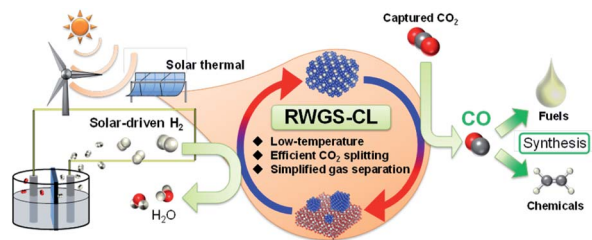


Fig. 1 Concept of solar-driven fuels and chemical synthesis processes with RWGS-CL.



Actually, RWGS-CL has benefits deriving from the absence of undesirable side reactions, which drives the equilibrium to CO production, and simplified gas separation, which leads to its improved energy efficiency. In fact, Wenzel *et al.* investigated solar-to-syngas efficiency for RWGS-CL processes and reported that the energy demand for separation is reduced by 77% compared to conventional RWGS processes.<sup>13</sup> In the RWGS-CL process, the development of the OSM is an indispensable factor to make this process feasible. The rates of reduction and oxidation, CO yield, and stability are emphasized as key factors affecting the OSM for the RWGS-CL cycles. Furthermore, the operation temperature is an important factor. Isothermal operation at lower temperatures is desired to reduce heat loss and to increase its feasibility.<sup>14</sup> In recent years, great effort has been put forth for the development of OSMs with high redox properties for RWGS-CL.<sup>14–19</sup> In previous studies, perovskite-type oxides<sup>14,15</sup> and iron-based oxides<sup>16–18</sup> have been well reported. Perovskite-type oxides are some of the promising OSMs thanks to their high stability and redox properties.<sup>20</sup> Maiti *et al.* investigated a series of lanthanum-based perovskites and found that  $\text{LaCa}_{0.4}\text{Fe}_{0.4}\text{Mn}_{0.6}\text{O}_3$  showed high  $\text{CO}_2$  splitting rate at low temperatures (723–823 K).<sup>15</sup> Iron-based oxides are other candidates of OSMs for RWGS-CL because of their high oxygen capacity. Cu, Co, and Mn co-doped ferrites<sup>16,17</sup> and  $\text{Fe}_2\text{O}_3$  supported on ion-conductive oxides<sup>18</sup> were developed, which showed high performance and stability for the RWGS-CL cycle at 923 K. However, in these OSMs, isothermal operations have been investigated at temperatures above 823 K, and the performance below 773 K is unclear. Although Utsis *et al.* reported isothermal RWGS-CL at 623–723 K using Fe-substituted Ba-hexa-aluminates,<sup>19</sup> the  $\text{CO}_2$  splitting capacity was  $0.4 \text{ mmol g}^{-1}$ , so higher performance of  $\text{CO}_2$  splitting is desired. This study has revealed that an indium-based oxide can be a novel OSM material able to exhibit extremely high isothermal RWGS-CL performance at low temperatures (673–773 K). Particularly, the performance of Cu-modified  $\text{In}_2\text{O}_3$  formed from  $\text{Cu}_2\text{In}_2\text{O}_5$  as a parent material was found to be promising for this purpose. Characterization of OSMs was conducted using powder X-ray diffraction (XRD), a field emission transmission electron microscope equipped with an energy-dispersive X-ray

spectrometer (STEM-EDX), X-ray photoelectron spectroscopy (XPS), and *in situ* X-ray absorption fine structure (XAFS) measurements.

## Results and discussion

### Isothermal RWGS-CL performance

After applying various oxides for RWGS-CL, we found that the Cu–In mixed oxide has high potential for RWGS-CL. The  $\text{CO}_2$  splitting performance obtained in this study is presented in Fig. 2 and Table S1† along with results described in earlier reports of the literature. Compared to other perovskite-type oxides<sup>14,15</sup> and ferrites,<sup>16,17</sup> which were reported earlier, Cu– $\text{In}_2\text{O}_3$  showed higher  $\text{CO}_2$  splitting rates and sufficient amounts of  $\text{CO}_2$  splitting even at low temperatures such as 673–773 K. Among all the oxides, Cu– $\text{In}_2\text{O}_3$  derived from  $\text{Cu}_2\text{In}_2\text{O}_5$  is the most promising material for low-temperature isothermal RWGS-CL. The cycling performance of isothermal RWGS-CL on  $\text{Cu}_2\text{In}_2\text{O}_5$  at 773 K is presented in ESI Fig. S1.† The reduction and oxidation (redox) amounts are defined as the moles of oxygen released or restored per gram of  $\text{Cu}_2\text{In}_2\text{O}_5$ . In the first cycle,  $\text{Cu}_2\text{In}_2\text{O}_5$  showed reduction of  $9.50 \text{ mmol g}^{-1}$  and re-oxidation of  $4.82 \text{ mmol g}^{-1}$ . Then, in cycles 2–5, the reduction and the oxidation amounts were, respectively,  $5.06 \pm 0.41 \text{ mmol g}^{-1}$  and  $4.80 \pm 0.40 \text{ mmol g}^{-1}$ . Excess reduction in the first cycle ( $4.68 \text{ mmol g}^{-1}$ ) was almost equivalent to the release of 2 mol oxygen atoms per mol of  $\text{Cu}_2\text{In}_2\text{O}_5$ . This finding suggests that  $\text{Cu}_2\text{In}_2\text{O}_5$  has become Cu(0)– $\text{In}_2\text{O}_3$ , which is confirmed later. Products of RWGS-CL were measured using a quadrupole mass spectrometer. The mass spectra of products are presented in ESI Fig. S2.† The products of reduction and oxidation steps were identified, respectively, as  $\text{H}_2\text{O}$  ( $m/z = 18$ ) and  $\text{CO}$  ( $m/z = 28$ ) without production of other by-products. Results confirmed that RWGS-CL on Cu– $\text{In}_2\text{O}_3$  formed from  $\text{Cu}_2\text{In}_2\text{O}_5$  is a selective  $\text{CO}_2$  conversion to CO. Structural characterization of this material was achieved using a multi-point BET method (results and discussion are presented in ESI Tables S2, S3 and ESI text†), powder XRD, STEM-EDX, and *in situ* XAFS measurements. The XRD patterns obtained from fresh and post-reaction samples are shown in Fig. 3A. The XRD pattern of the fresh sample showed the formation of  $\text{Cu}_2\text{In}_2\text{O}_5$  (ICDD 01-070-1082) with no

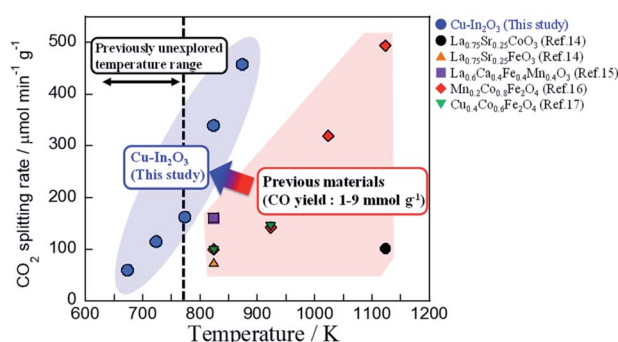


Fig. 2 Average  $\text{CO}_2$  splitting rates on the Cu–In mixed oxide (this study) and on various oxides in earlier reports.



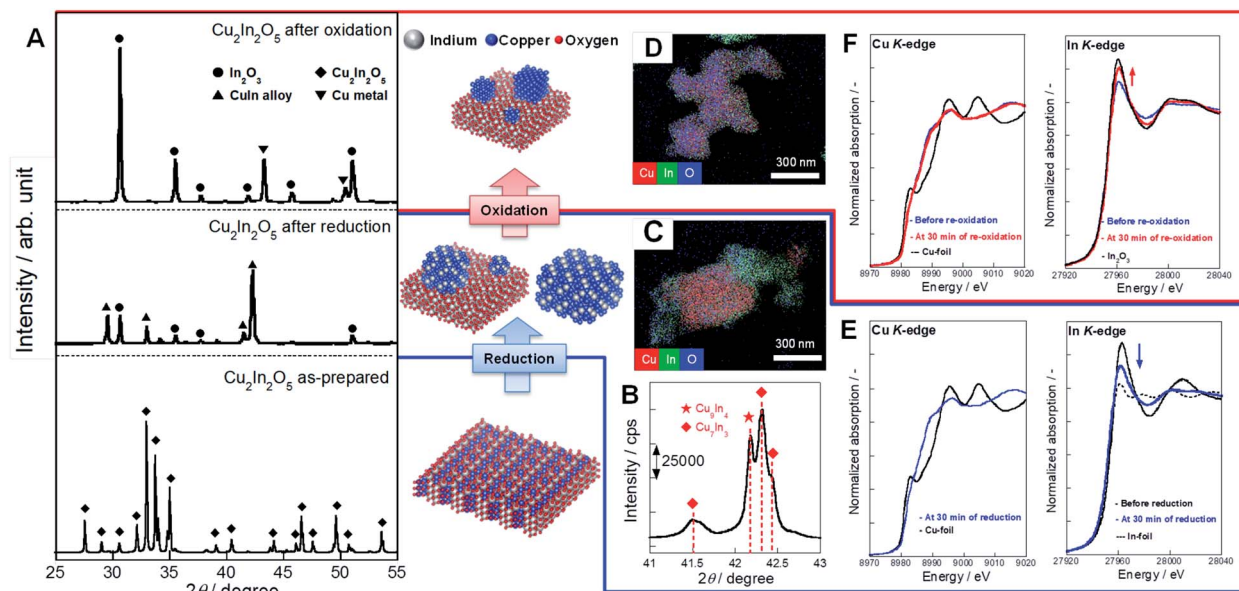


Fig. 3 Structural characterization of Cu–In<sub>2</sub>O<sub>3</sub> derived from Cu<sub>2</sub>In<sub>2</sub>O<sub>5</sub> during each step of the RWGS-CL cycle. (A) XRD diffraction patterns of Cu<sub>2</sub>In<sub>2</sub>O<sub>5</sub> as prepared, after reduction, and after oxidation. (B) Main diffraction peaks for the Cu–In alloy on the sample after reduction. EDX image of Cu–In<sub>2</sub>O<sub>3</sub> after reduction (C) and re-oxidation (D) at 773 K (copper, red; indium, green; oxygen, blue). Cu and In K-edge XANES spectra of Cu–In<sub>2</sub>O<sub>3</sub> during the reduction (E) and the re-oxidation (F).

impurity phase. The XRD pattern of the reduced sample exhibited diffraction peaks assigned to Cu–In alloy and In<sub>2</sub>O<sub>3</sub> (ICDD 01-089-4595). The main compositions of the alloy were Cu<sub>7</sub>In<sub>3</sub> (ICDD 03-065-2249) and Cu<sub>9</sub>In<sub>4</sub> (ICDD 00-042-1476) (Fig. 3B and S3†). From the phase diagram of the Cu–In system, these alloys are considered to be a solid phase during the RWGS-CL cycle.<sup>21</sup> Then, after the oxidation step, diffraction peaks of the Cu–In alloy disappeared. Those of Cu metal (ICDD 01-071-4610) and In<sub>2</sub>O<sub>3</sub> were observed. STEM-EDX images are presented in Fig. 3C, D and ESI Fig. S4.† The SEM micrographs of the reduced Cu–In<sub>2</sub>O<sub>3</sub> (ESI Fig. S4A†) show the presence of *ca.* 1 μm particles. The elemental composition of the particles was identified using EDX mapping (copper, red; indium, green; oxygen, blue). Results of EDX mapping of the sample after reduction (Fig. 3C and ESI Fig. S4A†) show that indium is distributed evenly over the particles, whereas copper is localized and oxygen is scattered. These images indicate Cu–In alloy formation by reduction. For the re-oxidized sample, many particles with uniformly distributed copper, indium, and oxygen were observed as depicted in Fig. 3D and ESI Fig. S4B.† Using image processing and analysis of these EDX images with Python (see the ESI text and Fig. S5 for details†), it was estimated that 84% of In was overlapping with Cu, and 93% of Cu was overlapping with In. These results demonstrate that a fine matrix of Cu–In–O was formed in the re-oxidized particles. The Cu in re-oxidized particles was regarded as highly distributed on the surface or incorporated into the In<sub>2</sub>O<sub>3</sub> structure. As additional information, *in situ* XAFS results are presented in Fig. 3E and F (see the ESI text, Table S4 and Fig. S6–S8 for details†). The sequential change of In K-edge XANES spectra indicates that the redox of In(III) ⇌ In(0) occurs during the RWGS-CL cycle. By contrast, the Cu K-edge XANES spectrum, which only slightly

changed during the RWGS-CL cycle, represented the formation of the Cu–In alloy.<sup>22,23</sup> This structural information obtained from XRD, STEM-EDX, and XAFS measurements demonstrates that the RWGS-CL cycle on Cu<sub>2</sub>In<sub>2</sub>O<sub>5</sub>, as a parent material, is based on the redox of indium with the formation and oxidation of the Cu–In alloy. This material during RWGS-CL is denoted hereinafter as Cu–In<sub>2</sub>O<sub>3</sub>.

To elucidate the role of Cu in Cu–In<sub>2</sub>O<sub>3</sub>, we compared the performance of isothermal RWGS-CL as a function of temperature on Cu–In<sub>2</sub>O<sub>3</sub> (derived from Cu<sub>2</sub>In<sub>2</sub>O<sub>5</sub>), 10 wt% Cu supported on In<sub>2</sub>O<sub>3</sub> (10 wt% Cu/In<sub>2</sub>O<sub>3</sub>) and pure In<sub>2</sub>O<sub>3</sub> without Cu, as shown in Fig. 4. Results indicate that Cu–In<sub>2</sub>O<sub>3</sub> is feasible for conducting RWGS-CL, even at 673 K. The performance at each temperature was much higher than that of 10 wt% Cu/In<sub>2</sub>O<sub>3</sub> and pure In<sub>2</sub>O<sub>3</sub> without Cu. Actually, the amount of redox on 10 wt% Cu/In<sub>2</sub>O<sub>3</sub> was slightly higher than that of In<sub>2</sub>O<sub>3</sub>,

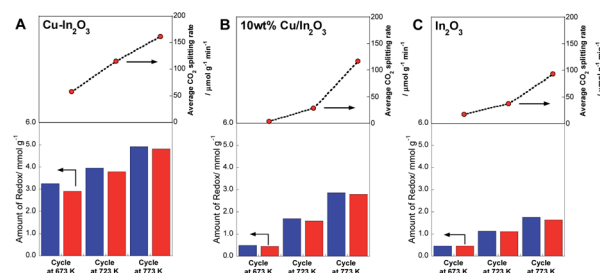


Fig. 4 Dependence of RWGS-CL performance of each oxide on the reaction temperature. Amounts of redox and average CO<sub>2</sub> splitting rate during the re-oxidation step on Cu–In<sub>2</sub>O<sub>3</sub> (A), 10 wt% Cu/In<sub>2</sub>O<sub>3</sub> (B), and In<sub>2</sub>O<sub>3</sub> (C). Reduction was conducted in a 10% H<sub>2</sub> atmosphere for 30 min; oxidation was conducted in a 10% CO<sub>2</sub> atmosphere.



although the CO<sub>2</sub> splitting rate was improved only slightly. Results of structural characterization of 10 wt% Cu/In<sub>2</sub>O<sub>3</sub> (ESI text and Fig. S9–S13† provide additional details) show that, because of the poor contact of Cu species with the In<sub>2</sub>O<sub>3</sub> surface, 10 wt% Cu/In<sub>2</sub>O<sub>3</sub> showed lower performance than Cu–In<sub>2</sub>O<sub>3</sub> derived from Cu<sub>2</sub>In<sub>2</sub>O<sub>5</sub>.

### Kinetics investigations of oxides during the RWGS-CL cycle

**Reduction step by H<sub>2</sub>.** Temperature-programmed reduction by H<sub>2</sub> (H<sub>2</sub>-TPR) measurements, the results of which are presented in Fig. 5A, were performed to investigate the reducibility of oxides. The ESI presents additional details.† For Cu-containing oxides (Cu–In<sub>2</sub>O<sub>3</sub> and 10 wt% Cu/In<sub>2</sub>O<sub>3</sub>), the H<sub>2</sub>-TPR profiles showed three notable reduction areas (*ca.* 400–550 K, 580–700 K, and 700–920 K). In the moderate temperature region of 580–700 K, the amount of reduction was equivalent to about 26% of oxygen contained in Cu<sub>2</sub>In<sub>2</sub>O<sub>5</sub>. Ploner *et al.* have observed that the Cu–In alloy formed at 640–723 K on the Cu–In<sub>2</sub>O<sub>3</sub> system in an H<sub>2</sub> atmosphere using *in situ* XRD measurement.<sup>24</sup> Therefore, the reduction of In(III) with the formation of the Cu–In alloy, the composition of which finally approached Cu<sub>9</sub>In<sub>4</sub>, was regarded as proceeding in this moderate temperature region. Regarding the results of H<sub>2</sub>-TPR measurements, results show that Cu species promote the reduction of indium oxide at low temperatures. To investigate more details of the mechanism, the kinetics model for the reduction of these oxides was verified using the method presented by Hancock and Sharp for the isothermal solid-state reaction.<sup>25–27</sup> The ESI text, Fig. S14–S17 and Tables S6–S8† present additional details. The kinetics models used for model fitting are shown in ESI Table S5.† As a result of the investigations, the phase-boundary-controlled reaction models (R2 and R3 models) were found to be suitable to describe the experimentally obtained data for all the oxides. These models are categorized as a classic shrinking core model in which the rate-controlling step is the surface chemical reaction.<sup>26,27</sup> The activation energy was investigated using the reaction rate constant *k* values obtained from the model fitting described above (ESI Table S9†). Plots of ln(*k*<sub>red</sub>/min<sup>-1</sup>) vs. 1/*T* (Arrhenius plot) for the three oxides are presented in Fig. 5B. The respective activation energies for Cu–In<sub>2</sub>O<sub>3</sub> (45.2 kJ mol<sup>-1</sup>) and 10 wt% Cu/In<sub>2</sub>O<sub>3</sub> (53.4 kJ mol<sup>-1</sup>) were much

lower than that for In<sub>2</sub>O<sub>3</sub> (76.6 kJ mol<sup>-1</sup>), confirming that the Cu species supported on In<sub>2</sub>O<sub>3</sub> decreased the activation barrier for reactions between the surfaces of In<sub>2</sub>O<sub>3</sub> and H<sub>2</sub>. The slight difference in the activation energy between Cu–In<sub>2</sub>O<sub>3</sub> and 10 wt% Cu/In<sub>2</sub>O<sub>3</sub> suggests that the effect of the Cu species on the reduction does not depend much on these morphologies.

**Oxidation step by CO<sub>2</sub> for CO formation.** Next, kinetics investigations were conducted for the oxidation step by CO<sub>2</sub> in the RWGS-CL cycle. The kinetics model for the oxidation of the oxide was also verified using the Hancock and Sharp method. As a consequence of the model fitting (details of which are presented in ESI Fig. S18–S21 and Tables S10–S12†), the most suitable models were concluded to be a zero-order model (R1 model)<sup>25–27</sup> for Cu–In<sub>2</sub>O<sub>3</sub> and the nucleation model, known as the Avrami–Erofe'ev model (AE1 model), for the other two oxides.<sup>28–30</sup> Fig. 6A presents the Arrhenius plot for the oxidation of these three oxides. The reaction rate constant *k* values obtained from model fitting are presented in Table S13.† The activation energy for Cu–In<sub>2</sub>O<sub>3</sub> is 50.7 kJ mol<sup>-1</sup>, which is markedly lower than that for either of the other two oxides. As the results of the model fitting show, the large difference in the activation energy is attributable to the different mechanisms of oxidation among these oxides. Considering that AE1 is suitable as a model for In<sub>2</sub>O<sub>3</sub> and 10 wt% Cu/In<sub>2</sub>O<sub>3</sub>, the overall oxidation of these two oxides is determined by the rate of formation and growth of the indium oxide nuclei. At the beginning of the reaction, the rate of oxidation is high because of random nucleation on the surface of the reduced oxide. However, as the nucleation and growth of nuclei proceed, the rate of oxidation plummets because of the decrease of the reduced surface. As a result, a long time is necessary to complete oxidation by CO<sub>2</sub>. Fig. 6B presents different rates of CO<sub>2</sub> splitting over these oxides during oxidation by CO<sub>2</sub>. For all oxides, the degree of reduction before re-oxidation was fixed at 3.0 mmol g<sup>-1</sup>. On In<sub>2</sub>O<sub>3</sub> and 10 wt% Cu/In<sub>2</sub>O<sub>3</sub>, the CO<sub>2</sub> splitting rate decreased drastically with the progress of oxidation. The oxidation of Cu–In<sub>2</sub>O<sub>3</sub> proceeded while maintaining a constant CO<sub>2</sub> splitting rate of about 340 μmol g<sup>-1</sup> min<sup>-1</sup> until conversion reached 60%. As a result, re-oxidation of the reduced Cu–In<sub>2</sub>O<sub>3</sub> is completed twice as rapidly as that of the other two oxides. These results obtained from the kinetics investigation of Cu–In<sub>2</sub>O<sub>3</sub> suggest that the amount of active sites for CO<sub>2</sub> splitting remains constant while oxidation is proceeding. The oxidation state of the Cu–In<sub>2</sub>O<sub>3</sub> surface during the oxidation step was analyzed by XPS. Compared with the In 3d<sub>5/2</sub> peak of fully re-oxidized Cu–In<sub>2</sub>O<sub>3</sub>, that of reduced Cu–In<sub>2</sub>O<sub>3</sub> was shifted to the lower binding energy (details in ESI Fig. S22 and ESI text†), which demonstrates the existence of In(0) species on the surface.<sup>31</sup> Fig. 6C shows the surface In/O ratio at each oxidation rate of the reduced Cu–In<sub>2</sub>O<sub>3</sub> and 10 wt% Cu/In<sub>2</sub>O<sub>3</sub>, as measured by XPS. The In/O ratio of the Cu–In<sub>2</sub>O<sub>3</sub> surface remained above 0.9, even at an oxidation rate of 75%, which indicates that bulk oxidation proceeds preferentially while the surface remains in a highly reduced state. This particular CO<sub>2</sub> splitting mechanism of Cu–In<sub>2</sub>O<sub>3</sub> is explainable as shown in Fig. 6D. First, CO<sub>2</sub> splits on the surface of the Cu–In alloy; CO and oxide ions (O<sup>2-</sup>) are generated. Subsequently, the O<sup>2-</sup> ions migrate to the bulk of the



Fig. 5 Kinetics analysis of each oxide in the reduction step. (A) H<sub>2</sub>-TPR profiles for Cu–In<sub>2</sub>O<sub>3</sub> (derived from Cu<sub>2</sub>In<sub>2</sub>O<sub>5</sub>), 10 wt% Cu/In<sub>2</sub>O<sub>3</sub> and In<sub>2</sub>O<sub>3</sub>. (B) Arrhenius plots for the reduction of Cu–In<sub>2</sub>O<sub>3</sub> (derived from Cu<sub>2</sub>In<sub>2</sub>O<sub>5</sub>), 10 wt% Cu/In<sub>2</sub>O<sub>3</sub> and In<sub>2</sub>O<sub>3</sub>.





Fig. 6 Kinetics analysis of each oxide in re-oxidation steps. (A) Arrhenius plots for the oxidation of Cu–In<sub>2</sub>O<sub>3</sub> (derived from Cu<sub>2</sub>In<sub>2</sub>O<sub>5</sub>), 10 wt% Cu/In<sub>2</sub>O<sub>3</sub> and In<sub>2</sub>O<sub>3</sub>. (B) Dependence of differential CO<sub>2</sub> splitting rate on conversion of oxides. (C) Surface In/O ratios on Cu–In<sub>2</sub>O<sub>3</sub> and 10 wt% Cu/In<sub>2</sub>O<sub>3</sub> during the re-oxidation step. (D) Presumed mechanism of CO<sub>2</sub> splitting on the reduced Cu–In<sub>2</sub>O<sub>3</sub>.

particle. They presumably oxidize the Cu–In alloy–In<sub>2</sub>O<sub>3</sub> interface. As a result, the alloy surface, as the active site, remains in a highly reduced state. Rapid CO<sub>2</sub> splitting proceeds continuously.

## Conclusion

As reported herein, Cu–In<sub>2</sub>O<sub>3</sub> was synthesized. Then its performance for RWGS-CL was investigated at the low temperatures of 673–773 K. Results of RWGS-CL cycle tests demonstrated that, even at low temperatures, Cu–In<sub>2</sub>O<sub>3</sub> exhibited much higher CO<sub>2</sub> splitting performance than ever reported. Results of XRD, STEM-EDX, and *in situ* XAFS measurements showed that the oxide has a structure of reduced Cu supported on In<sub>2</sub>O<sub>3</sub> (Cu–In<sub>2</sub>O<sub>3</sub>) under the RWGS-CL conditions, and showed that its redox properties are based on Cu–In alloy formation and re-oxidation. In addition, because low performance was exhibited by pure In<sub>2</sub>O<sub>3</sub> and Cu supported In<sub>2</sub>O<sub>3</sub> that were prepared using an impregnation method, the structured state of Cu–In<sub>2</sub>O<sub>3</sub> has extremely important roles in its high RWGS-CL performance. H<sub>2</sub>-TPR and kinetics investigations revealed that Cu–In<sub>2</sub>O<sub>3</sub> shows high reducibility, even at low temperatures. We concluded that Cu species promote the reaction of H<sub>2</sub> with the oxide surface. Results of kinetics investigations in the oxidation step indicate that the Cu–In alloy particle surface conserved a highly reduced state even under

oxidation condition, resulting in rapid completion of re-oxidation by CO<sub>2</sub>. The interesting oxidation behavior is attributed to rapid O<sup>2–</sup> migration from the surface to the bulk of the Cu–In alloy and preferential oxidation of the interface of alloy–In<sub>2</sub>O<sub>3</sub>. The combination of high reducibility and the specific re-oxidation mechanism engenders the high performance found for RWGS-CL on Cu–In<sub>2</sub>O<sub>3</sub>.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

*In situ* XAFS measurements were performed at the beamline BL07 of the SAGA Light Source (Proposal No. 1910093P/BL07). The authors thank Dr H. Setoyama for help with measurements of XANES spectra.

## Notes and references

- IPCC, [www.ipcc.ch/site/assets/uploads/2018/02/SYR\\_AR5\\_FINAL\\_full.pdf](http://www.ipcc.ch/site/assets/uploads/2018/02/SYR_AR5_FINAL_full.pdf).
- S. J. Davis, K. Caldeira and H. D. Matthews, *Science*, 2010, **329**, 1330–1333.



- 3 S. Perathoner and G. Centi, *ChemSusChem*, 2014, **7**, 1274–1282.
- 4 J. Artz, T. E. Müller, K. Thenert, J. Kleinekorte, R. Meys, A. Sternberg, A. Bardow and W. Leitner, *Chem. Rev.*, 2018, **118**, 434–504.
- 5 D. S. Mallapragada, N. R. Singh, V. Curteanu and R. Agrawal, *Ind. Eng. Chem. Res.*, 2013, **52**, 5136–5144.
- 6 F. V. Vázquez, J. Koponen, V. Ruuskanen, C. Bajamundi, A. Kosonen, P. Simell, J. Ahola, C. Frilund, J. Elfving, M. Reinikainen, N. Heikkinen, J. Kauppinen and P. Piermartini, *J. CO<sub>2</sub> Util.*, 2018, **28**, 235–246.
- 7 I. Dimitriou, P. García-Gutiérrez, R. H. Elder, R. M. Cuéllar-Franca, A. Azapagic and R. W. K. Allen, *Energy Environ. Sci.*, 2015, **8**, 1775–1789.
- 8 M. Romero and A. Steinfeld, *Energy Environ. Sci.*, 2012, **5**, 9234–9245.
- 9 P. Kaiser, R. B. Unde, C. Kern and A. Jess, *Chem. Ing. Tech.*, 2013, **85**, 489–499.
- 10 Y. A. Daza and J. N. Kuhn, *RSC Adv.*, 2016, **6**, 49675–49691.
- 11 K. Oshima, T. Shinagawa, Y. Nogami, R. Manabe, S. Ogo and Y. Sekine, *Catal. Today*, 2016, **232**, 27–32.
- 12 S.-C. Yang, S. H. Pang, T. P. Sulmonetti, W.-N. Su, J.-F. Lee, B.-J. Hwang and C. W. Jones, *ACS Catal.*, 2018, **8**, 12056–12066.
- 13 M. Wenzel, L. Rihko-Struckmann and K. Sundmacher, *AIChE J.*, 2017, **63**, 15–22.
- 14 Y. A. Daza, D. Maiti, R. A. Kent, V. R. Bhethanabotla and J. N. Kuhn, *Catal. Today*, 2015, **258**, 691–698.
- 15 D. Maiti, B. J. Hare, Y. A. Daza, A. E. Ramos, J. N. Kuhn and V. R. Bhethanabotla, *Energy Environ. Sci.*, 2018, **11**, 648–659.
- 16 L. Ma, Y. Qiu, M. Li, D. Cui, S. Zhang, D. Zheng and R. Xiao, *Ind. Eng. Chem. Res.*, 2020, **59**, 6924–6930.
- 17 Y. Qiu, L. Ma, D. Zeng, M. Li, D. Cui, Y. Lv, S. Zhang and R. Xiao, *J. Energy Chem.*, 2020, **46**, 123–132.
- 18 D. Zeng, Y. Qiu, L. Ma, M. Li, D. Cui, S. Zhang and R. Xiao, *Environ. Sci. Technol.*, 2020, **54**, 12467–12475.
- 19 N. Utsis, M. V. Landau, A. Erenburg and M. Herskowitz, *Catalysts*, 2020, **10**, 1082.
- 20 X. Zhu, K. Li, L. Neal and F. Li, *ACS Catal.*, 2018, **8**, 8213–8236.
- 21 Z. Bahari, E. Dichi, B. Legendre and J. Dugué, *Thermochim. Acta*, 2003, **401**, 131–138.
- 22 M. G. Kim, S. Sim and J. Cho, *Adv. Mater.*, 2010, **22**, 5154–5158.
- 23 W.-R. Lee, M. G. Kim, J.-R. Choi, J.-I. Park, S. J. Ko, S. J. Oh and J. Cheon, *J. Am. Chem. Soc.*, 2005, **127**, 16090–16097.
- 24 K. Ploner, L. Schlicker, A. Gili, A. Gurlo, A. Doran, L. Zhang, M. Armbrüster, D. Obendorf, J. Bernardi, B. Klötzer and S. Penner, *Sci. Technol. Adv. Mater.*, 2019, **20**, 356–366.
- 25 J. D. Hancock and J. H. Sharp, *J. Am. Ceram. Soc.*, 1972, **55**, 74–77.
- 26 K. Piotrowski, K. Mondal, H. Lorethova, L. Stonawski, T. Szymański and T. Wiltowski, *Int. J. Hydrogen Energy*, 2005, **30**, 1543–1554.
- 27 Z. Zhou, L. Han and G. M. Bollas, *Int. J. Hydrogen Energy*, 2014, **39**, 8535–8556.
- 28 M. Avrami, *J. Chem. Phys.*, 1939, **7**, 1103–1112.
- 29 M. Avrami, *J. Chem. Phys.*, 1940, **8**, 212–224.
- 30 M. Avrami, *J. Chem. Phys.*, 1941, **9**, 177–184.
- 31 J. Gao, F. Song, Y. Li, W. Cheng, H. Yuan and Q. Xu, *Ind. Eng. Chem. Res.*, 2020, **59**, 12331–12337.

