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## Divalent transition metal ion intercalated hydrated V<sub>2</sub>O<sub>5</sub> nanosheet cathodes for ultra-long cycling aqueous zinc-ion batteries†

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Given the rich d-electron structures of transition metals, we synthesized Co<sup>2+</sup>/Ni<sup>2+</sup>/Cu<sup>2+</sup>-intercalated hydrated V<sub>2</sub>O<sub>5</sub> nanosheet cathodes via a novel self-assembly approach, achieving excellent performance in AZIBs. DFT calculations revealed that the intercalation of transition metal ions modulates the electronic structure near the Fermi level of pristine V<sub>2</sub>O<sub>5</sub>, increasing DOS and reducing the band gap, thereby improving electrical conductivity.

The accelerated advancements in large-scale energy storage and flexible electronic devices have spurred a substantial demand for sophisticated rechargeable battery systems beyond lithium-ion technology.<sup>1,2</sup> Aqueous zinc-ion batteries (AZIBs) have garnered considerable attention among the next generation of energy storage units,<sup>3–5</sup> owing to the impressive theoretical specific capacities (820 mA h g<sup>−1</sup> and 5854 mA h cm<sup>−3</sup>) and low redox potential (−0.763 V *versus* the standard hydrogen electrode) of the metallic zinc anode, as well as the enhanced safety and cost-effectiveness of AZIBs.<sup>6–9</sup> Vanadium-based compounds have garnered significant attention due to their remarkable advantages in specific capacity, positioning them as promising candidates for cathode materials in next-generation large-scale AZIB energy storage systems.<sup>4,10</sup> Moreover, the work of Yuan *et al.* has substantially advanced and broadened the systematic investigation of cathode modification and anode protection in AZIBs.<sup>11–13</sup>

Layered materials have been extensively recognized in the field of electrochemical energy storage due to their abundant and tunable intercalation chemistry, as well as their high active surface.<sup>14–17</sup> In addition, their two-dimensional (2D) nanosheets, with thicknesses ranging from monolayer to multilayer, offer

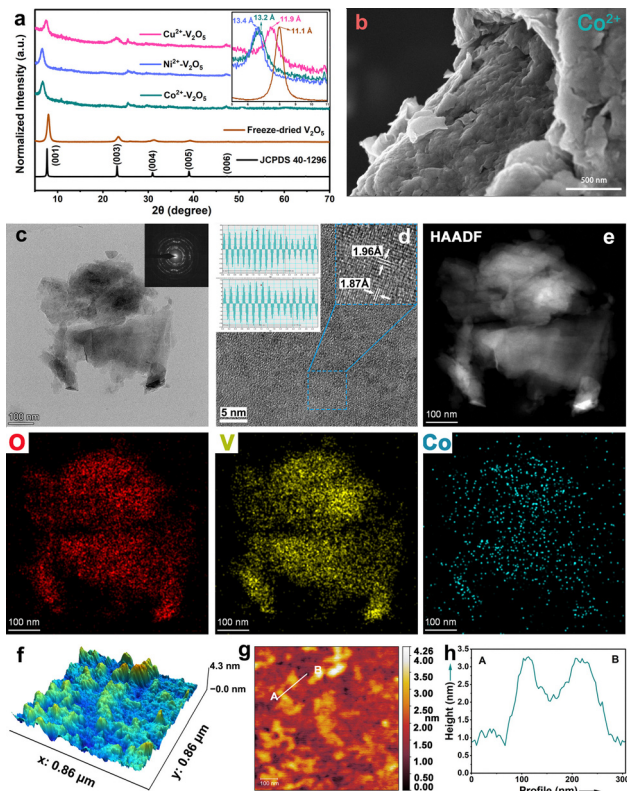
broad opportunities for subsequent structural and functional modifications.<sup>18</sup> Vanadium pentoxide (V<sub>2</sub>O<sub>5</sub>), as a representative layered vanadium oxide, possesses a high theoretical specific capacity of 589 mA h g<sup>−1</sup> due to its variable valence states (from 5+ to 3+), which renders it widely utilized as a cathode material in AZIBs.<sup>19</sup> Research indicates that metal cations embedded into the layered structure of vanadium pentoxide serve as effective pillars within the structural framework, while also expanding the interlayer spacing and achieving exceptional zinc ion transport kinetics and storage capabilities.<sup>20</sup> Additionally, the incorporation of metal cations can effectively modulate the electronic structure near the Fermi level of pristine V<sub>2</sub>O<sub>5</sub>, enhancing its electrical conductivity, thereby improving electrochemical performance.<sup>21,22</sup> Notably, as intercalation ion candidates, transition metal cations have garnered significant attention due to their rich d-electron structures.<sup>23,24</sup> For instance, Bai *et al.* synthesized Cu<sup>2+</sup> and Mn<sup>2+</sup> co-intercalated hydrated V<sub>2</sub>O<sub>5</sub> cathode materials, which exhibited superior Zn<sup>2+</sup> diffusion kinetics and long cycling stability in AZIBs.<sup>23</sup> Herein, we demonstrate a hydrosol ion-mediated self-assembly strategy enabling the successful intercalation of three transition metal cations (Co<sup>2+</sup>, Ni<sup>2+</sup>, and Cu<sup>2+</sup>) into the layered architecture of a hydrated V<sub>2</sub>O<sub>5</sub> nanosheet, while preserving their structural integrity and ion-transport functionality. Moreover, compared to hydrothermal/solvothermal synthesis approaches, this self-assembly synthesis strategy enables the formation of a uniform phase structure.<sup>20</sup>

X-ray diffraction (XRD) patterns of the Co<sup>2+</sup>, Ni<sup>2+</sup>, and Cu<sup>2+</sup> assembling products are given in Fig. 1a. The sharp peaks near 5–9° can be identified as the (001) characteristic peaks. XRD analyses reveal that, compared to pure V<sub>2</sub>O<sub>5</sub> prepared by freeze-drying, the incorporation of Co<sup>2+</sup> (13.2 Å), Ni<sup>2+</sup> (13.4 Å), and Cu<sup>2+</sup> (11.9 Å) ions induces a leftward shift of the (001) characteristic peak, indicating an expansion of the interlayer spacing. Fourier transform infrared (FTIR) and Raman spectra also confirmed the successful synthesis of the three products (Fig. S1 and S2, ESI†). X-ray photoelectron spectroscopy (XPS) analyses further confirmed the successful intercalation of Co<sup>2+</sup>,

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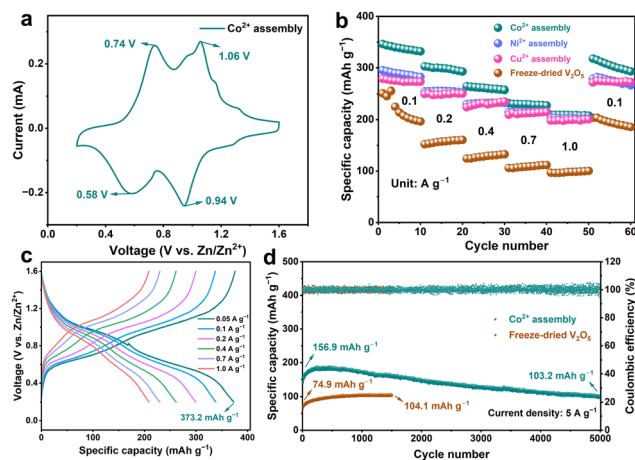


**Fig. 1** (a) XRD patterns of the freeze dried  $V_2O_5$  and  $Co^{2+}$ ,  $Ni^{2+}$ , and  $Cu^{2+}$ , cooperative self-assembly products, respectively. (b) The FESEM image, (c) TEM image and corresponding selected area electron diffraction (SAED) pattern, (d) HRTEM image, (e) EDS mapping images, (f) and (g) AFM images and (h) corresponding height profiles of the  $Co^{2+}$  self-assembly sample.

$Ni^{2+}$ , and  $Cu^{2+}$  (Fig. S3, ESI<sup>†</sup>). The field emission scanning electron microscopy (FESEM) and environmental scanning electron microscopy (ESEM) observations of the self-assembled samples reveal that their microscopic morphology exhibits large aggregates formed by the stacking of  $V_2O_5$  nanosheets (Fig. 1b and Fig. S4, ESI<sup>†</sup>). The transmission electron microscopy (TEM) image of the  $Co^{2+}$  intercalated product (Fig. 1c) also reveals a two-dimensional nanosheet morphology. The high-resolution transmission electron microscopy (HRTEM) image captured from the surface of the nanosheets in Fig. 1d reveals well-defined 2D lattice fringes, with spacings of 0.187 and 0.196 nm corresponding to the interlayer spacings of the (006) planes of layered  $V_2O_5$ . The energy dispersive spectrum (EDS) mapping (Fig. 1e) demonstrates the uniform distribution of V, O, and Co elements within the sample, while the presence of Co distribution confirms the successful intercalation of  $Co^{2+}$  into the layered  $V_2O_5 \cdot nH_2O$  host. Atomic force microscopy (AFM) characterization revealed that the self-assembled  $Co^{2+}$  product exhibits a thickness ranging from 2.0 to 3.5 nm, confirming its two-dimensional stacked nanosheet structure, which is consistent with the findings from SEM analysis (Fig. 1f–h). The chemical formula of pristine  $V_2O_5 \cdot nH_2O$  and these samples intercalated by  $Co^{2+}$ ,  $Ni^{2+}$ , and  $Cu^{2+}$  was determined by inductive coupled plasma (ICP) analysis (Table S1, ESI<sup>†</sup>)

and thermogravimetric analysis (TGA) (Fig. S5, ESI<sup>†</sup>), which can be identified as  $V_2O_5 \cdot 0.77H_2O$ ,  $Co_{0.14}V_2O_5 \cdot 0.95H_2O$ ,  $Ni_{0.15}V_2O_5 \cdot 0.63H_2O$ , and  $Cu_{0.15}V_2O_5 \cdot 0.60H_2O$ , respectively.

Subsequently, the self-assembled samples were employed as cathode materials to assemble AZIBs using a 2 M  $Zn(OTf)_2$  electrolyte and a zinc foil. According to a previous study,<sup>20</sup> the specific capacity of self-assembled products is highly dependent on the aging time following the self-assembly process. Therefore, we systematically investigated the impact of aging time after the  $Co^{2+}$  self-assembly process and conducted electrochemical evaluations. Galvanostatic charge–discharge (GCD) testing at  $0.05 A g^{-1}$  revealed that the sample aged for 5 h exhibited the optimal capacity performance, further confirming the critical influence of aging time on the specific capacity (Fig. S6, ESI<sup>†</sup>). Consequently, an aging time of 5 h was ultimately adopted for all self-assembled products. Cyclic voltammetry (CV) tests demonstrated that  $Co_{0.14}V_2O_5 \cdot 0.95H_2O$ ,  $Ni_{0.15}V_2O_5 \cdot 0.63H_2O$  and  $Cu_{0.15}V_2O_5 \cdot 0.60H_2O$  all exhibited pronounced redox peaks, indicative of typical battery behavior (Fig. 2a, and Fig. S7, ESI<sup>†</sup>). Compared to the pristine  $V_2O_5$  prepared by direct freeze-drying, the rate performance tests of  $Co^{2+}$ ,  $Ni^{2+}$ , and  $Cu^{2+}$  self-assembled samples exhibit superior discharge specific capacities (Fig. 2b). This enhancement is likely attributed to the modulation of the electronic structure of the pristine  $V_2O_5$  by the rich d-electron of the transition metals, leading to improved conductivity. The  $Co_{0.14}V_2O_5 \cdot 0.95H_2O$  cathode exhibited the highest discharge-specific capacity, which achieved average specific capacities of 338.2, 298.3, 260.8, 229.5, and 208.8  $mA h g^{-1}$  at current densities of 0.1, 0.2, 0.4, 0.7, and  $1.0 A g^{-1}$ , respectively. When the current density was reset to  $0.1 A g^{-1}$ , the specific capacity recovered to  $304.7 mA h g^{-1}$ . Notably, the  $Co_{0.14}V_2O_5 \cdot 0.95H_2O$  cathode exhibited an outstanding discharge specific capacity of  $373.2 mA h g^{-1}$  at a current density of  $0.05 A g^{-1}$  (Fig. 2c). The electrochemical performance outlined above still presents certain advantages compared to previously



**Fig. 2** (a) CV profile. (b) The corresponding rate performance profiles of  $Co^{2+}$ ,  $Ni^{2+}$  and  $Cu^{2+}$  cathodes. (c) The corresponding GCD profiles at various current densities of the  $Co_{0.14}V_2O_5 \cdot 0.95H_2O$  cathode. (d) Long-term cycling stability at  $5.0 A g^{-1}$  of  $Co_{0.14}V_2O_5 \cdot 0.95H_2O$  and freeze-dried  $V_2O_5$  cathodes.

published studies (Table S2, ESI†). The assembled  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$ ,  $\text{Ni}_{0.15}\text{V}_2\text{O}_5 \cdot 0.63\text{H}_2\text{O}$ , and  $\text{Cu}_{0.15}\text{V}_2\text{O}_5 \cdot 0.60\text{H}_2\text{O}$  cathodes were subjected to high-current long-term cycling performance tests (Fig. 2d, and Fig. S8, ESI†). It is noteworthy that the  $\text{Co}^{2+}$  self-assembled sample demonstrates superior cycling performance. The  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  cathode retains a discharge-specific capacity of  $103.2 \text{ mA h g}^{-1}$  even after 5000 cycles at a current density of  $5 \text{ A g}^{-1}$ . The excellent cycling stability under high current density could be attributed to the appropriate interlayer spacing adjustment induced by  $\text{Co}^{2+}$  intercalation, as well as the pillaring effect between  $\text{Co}^{2+}$  and the host frame.<sup>20</sup> The GCD curves and low-current cycling performance profiles of the  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  electrode further exemplify this characteristic (Fig. S9, ESI†).

A series of electrochemical evaluations confirmed that the  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  cathode exhibits the optimal electrochemical performance among the self-assembled products derived from the three transition metal cations. Compared with  $\text{Ni}^{2+}$  and  $\text{Cu}^{2+}$ ,  $\text{Co}^{2+}$  demonstrates superior electrochemical activity and stability due to its unique d-electron configuration, appropriate electrochemical potential, and robust Co–O bonds. These attributes contribute to more efficient charge transfer processes, thereby affording enhanced electrochemical performance. To further investigate the zinc ion storage and diffusion mechanisms of this self-assembly series, kinetic analyses were conducted. The CV curves of the  $\text{Zn}/\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  battery were measured at different scan rates (Fig. 3a and Notes S1, ESI†). As shown in Fig. 3b, the  $b$  values of peaks 1 to 4 are 0.792, 0.791, 0.790, and 0.935, respectively, indicating that the storage of  $\text{Zn}^{2+}$  is predominantly governed by capacitive contributions. CV measurements at various scan rates indicate that the contribution of capacitive behavior becomes increasingly dominant with the increase in scan rate (Fig. 3c and d and Fig. S10, Notes S2, ESI†). The above results demonstrate that the  $\text{Zn}^{2+}$  storage kinetics in the  $\text{Zn}/\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  battery are predominantly governed by capacitive behavior. In addition,

electrochemical impedance spectroscopy (EIS) measurements were conducted to elucidate the charge transfer resistance ( $R_{\text{ct}}$ ). The  $R_{\text{ct}}$  values of  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$ ,  $\text{Ni}_{0.15}\text{V}_2\text{O}_5 \cdot 0.63\text{H}_2\text{O}$ , and  $\text{Cu}_{0.15}\text{V}_2\text{O}_5 \cdot 0.60\text{H}_2\text{O}$  are 92.7, 111.3 and  $74.3 \text{ } \Omega$ , respectively, all of which are notably lower than that of the freeze-dried pristine sample ( $\text{V}_2\text{O}_5 \cdot 0.77\text{H}_2\text{O}$ ,  $225.1 \text{ } \Omega$ ) (Fig. 3e). Notably, the average ion diffusion coefficient ( $\bar{D}_{\text{Zn}^{2+}}$ ) value of  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$ , as determined by the galvanostatic intermittent titration technique (GITT), is higher than those of  $\text{Ni}_{0.15}\text{V}_2\text{O}_5 \cdot 0.63\text{H}_2\text{O}$ ,  $\text{Cu}_{0.15}\text{V}_2\text{O}_5 \cdot 0.60\text{H}_2\text{O}$ , and freeze-dried  $\text{V}_2\text{O}_5$  (Fig. 3f and Fig. S11, Notes S3, ESI†). These results are consistent with the aforementioned unique electronic structure of  $\text{Co}^{2+}$ .

The density functional theory (DFT) calculation results indicated that the insertion of  $\text{Co}^{2+}$  modulates the electronic structure of pristine  $\text{V}_2\text{O}_5$ , reducing its band gap and enhancing its electrical conductivity (Fig. S12, ESI†). This leads to superior charge transfer kinetics in  $\text{Co}^{2+}$ -intercalated  $\text{V}_2\text{O}_5$ . The EIS measurements of  $\text{V}_2\text{O}_5 \cdot 0.77\text{H}_2\text{O}$  and  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  further corroborated these computational findings. *Ex situ* XRD characterization and XPS spectra further confirmed the typical  $\text{Zn}^{2+}$  intercalation/de-intercalation mechanism of the  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  cathode, which is consistent with that of most previously reported vanadium-based cathodes (Fig. S13, ESI†). In summary, the electrochemical reaction mechanism of  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  as a cathode for AZIBs can be delineated as shown in Fig. S14 (ESI†). The anodic reaction of the  $\text{Zn}/\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  cell is a typical process of metallic zinc losing electrons ( $\text{Zn} \rightarrow \text{Zn}^{2+} + \text{xe}^-$ ). We further performed ESEM analysis on different states of the first GCD cycle of the  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  cathode at a current density of  $0.1 \text{ A g}^{-1}$  (Fig. S15, ESI†). The formation of products distinct from the original cathode material at different discharge states is also consistent with the aforementioned *ex situ* XRD and XPS characterization results. Furthermore, the investigations of *in situ* optical microscopy (OM) unambiguously corroborated the exceptional structural stability of the  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  electrode, as evidenced by its preserved dimensional integrity throughout prolonged electrochemical cycling under intercalation/de-intercalation conditions (Fig. 4, and Notes S4, ESI†).

In conclusion, using a novel hydrosol ion-mediated self-assembly strategy, we successfully intercalated three transition metal cations ( $\text{Co}^{2+}$ ,  $\text{Ni}^{2+}$ , and  $\text{Cu}^{2+}$ ) into the layered framework of hydrated vanadium pentoxide ( $\text{V}_2\text{O}_5 \cdot n\text{H}_2\text{O}$ ) nanosheets. The self-assembled product of  $\text{Co}^{2+}$  demonstrates superior electrochemical performance, outperforming the self-assembled products of  $\text{Ni}^{2+}$  and  $\text{Cu}^{2+}$ , which can be attributed to its high electrochemical activity and excellent structural stability. DFT calculation results indicated that  $\text{Co}^{2+}$ -intercalated  $\text{V}_2\text{O}_5$  exhibits an enhanced electronic DOS near the Fermi level and a reduced band gap value, enhancing the electrical conductivity of pristine  $\text{V}_2\text{O}_5$ . Electrochemical testing results demonstrated that the  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  cathode retains a high discharge specific capacity of  $103.2 \text{ mA h g}^{-1}$  even after 5,000 cycles at a high current density of  $5 \text{ A g}^{-1}$ . Our work provides valuable insights into the development of high-performance vanadium-based cathode materials for aqueous batteries.

Jie Bai: writing – original draft, software, methodology, investigation, formal analysis, and data curation. Wenting Ji:



**Fig. 3** (a) CV curves at various scan rates, (b) the as-calculated  $b$  value of four different peaks 1–4, (c) capacitive and diffusion contribution bar graph calculated from the dependence of peak current density and the scan rate, and (d) CV curves containing the schematic of capacitive contribution at a current density of  $1.0 \text{ mV s}^{-1}$ , for the  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  cathode. (e) EIS spectrum of freeze-dried  $\text{V}_2\text{O}_5$ ,  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$ ,  $\text{Ni}_{0.15}\text{V}_2\text{O}_5 \cdot 0.63\text{H}_2\text{O}$  and  $\text{Cu}_{0.15}\text{V}_2\text{O}_5 \cdot 0.60\text{H}_2\text{O}$  cathodes. (f) GITT analysis of the  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  electrode.



Fig. 4 The *in situ* OM characterization of the  $\text{Co}_{0.14}\text{V}_2\text{O}_5 \cdot 0.95\text{H}_2\text{O}$  cathode. (a)–(j) correspond to discharge to 0.2 V, while (k)–(p) corresponds to charge to 1.6 V.

data curation, software, and formal analysis. Mengda Xue: data curation. De Li: data curation and software. Huayu Wang: data curation and analysis. Lingyun Chen: conceptualization, funding acquisition, supervision, validation, writing – review and editing.

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## Conflicts of interest

There are no conflicts to declare.

## Data availability

The data that support the findings of this study are available in the ESI.†

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