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Topological data analysis enhanced prediction of hydrogen storage in metal–organic frameworks (MOFs)[†]

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Metal–organic frameworks (MOFs) have the capacity to serve as gas capturing, sensing, and storing systems. It is usual practice to select the MOF from a vast database with the best adsorption property in order to do an adsorption calculation. The costs of computing thermodynamic values are sometimes a limiting factor in high-throughput computational research, inhibiting the development of MOFs for separations and storage applications. In recent years, machine learning has emerged as a promising substitute for traditional methods like experiments and simulations when trying to foretell material properties. The most difficult part of this process is choosing characteristics that produce interpretable representations of materials that may be used for a variety of prediction tasks. We investigate a feature-based representation of materials using tools from topological data analysis. In order to describe the geometry of MOFs with greater accuracy, we use persistent homology. We show our method by forecasting the hydrogen storage capacity of MOFs during a temperature and pressure swing from 100 bar/77 K to 5 bar/160 K, using the synthetically compiled CoRE MOF-2019 database of 4029 MOFs. Our topological descriptor is used in conjunction with more conventional structural features, and their usefulness to prediction tasks is explored. In addition to demonstrating significant progress over the baseline, our findings draw attention to the fact that topological features capture information that is supplementary to the structural features.

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1 Introduction

The push to create new fuel sources was sparked by the depleting supply of fossil fuels and climate change brought on by carbon dioxide emissions. When looking to phase out carbon-based energy sources, hydrogen (H₂) is among the most attractive alternatives. H₂ is widely considered to be the clean, sustainable fuel of the future because of its potential to replace fossil fuels in the energy sector and its many advantages over traditional energy sources, including its abundance, low environmental impact during combustion, and high specific energy.^{1–3} H₂ gas's extremely low volumetric energy density due to its volatility to ambient conditions is a major roadblock to its broad usage as a primary fuel source, especially for on-board mobile transportation.^{4,5} As a result, there is still a lot of focus on making sure H₂ is stored in a way that is safe, efficient,

and technically and economically viable.^{6,7} Adsorption as a method for enhancing H₂ storage density has been presented and has garnered a lot of interest.^{8–10} However, the technique will only be successful if new materials are developed that can store a substantial quantity of H₂ under mild conditions, while also being compact, lightweight, having fast kinetics for charging and delivering H₂, and being highly reversible.

Since metal–organic frameworks (MOFs) are synthesised modularly from metal centres and organic ligands, they are a novel class of functional porous crystalline solids with a wide range of controllable properties and a wide variety of chemical and structural forms.^{11,12} Theoretically, MOF materials can be designed in an almost infinite number of ways, with the modification of metal ions/clusters and organic ligands enabling for the tailoring of their porosity and pore chemistry for an extensive range of conceivable uses.^{13,14} For a variety of adsorbent applications, MOFs have shown promise.^{15,16} The BET surface area and porosity of MOFs are relatively large in contrast to other materials that may store hydrogen molecules, such as CNT, hydrides, zeolites, and clathrates. Opportunities in gas separation, catalysis, energy storage, and conversion have recently piqued interest in MOF-based materials. Predicting the adsorption and diffusion behaviour of guest species in

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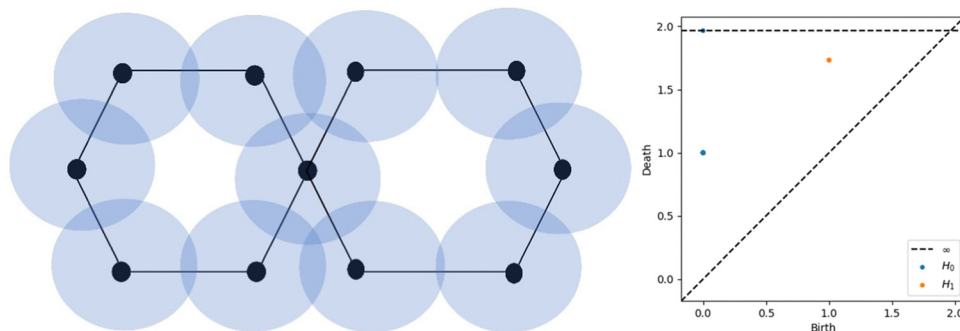


Fig. 1 Schematic representation of point cloud to persistence diagram.

employ a modified version of the PersIm package (<https://persim.scikit-tda.org>). By integrating the resulting mixture of Gaussians in the cells of the grid, the birth vs. persistence points are discretized onto a grid of fixed size. We do this with a 50×50 pixel grid and a Gaussian blur of $\sigma = 0.15$ to accomplish the desired effect as discussed in the previous papers.^{30,31} Fig. 2 represents an example MOF structure and then the corresponding PD and PI are shown in Fig. 2 and 3.

2.4 ResNet model

Residual networks (or ResNet for short) was coined by He *et al.* in 2015.³⁶ The ResNet architecture was designed to overcome

the problem of vanishing gradients in deep neural networks. This problem occurs when the gradients become too small during backpropagation, which can lead to slower convergence or even complete saturation of the network. ResNet uses skip connections, also known as residual connections, to address this issue. These connections allow the network to skip one or more layers and pass the input directly to a later layer, which helps to preserve the gradient and prevent it from becoming too small.

The 72-layer architecture used in ResNet-18 is 18 layers deep. This network was built with the intention of efficiently supporting many convolutional layers with varying filter sizes and

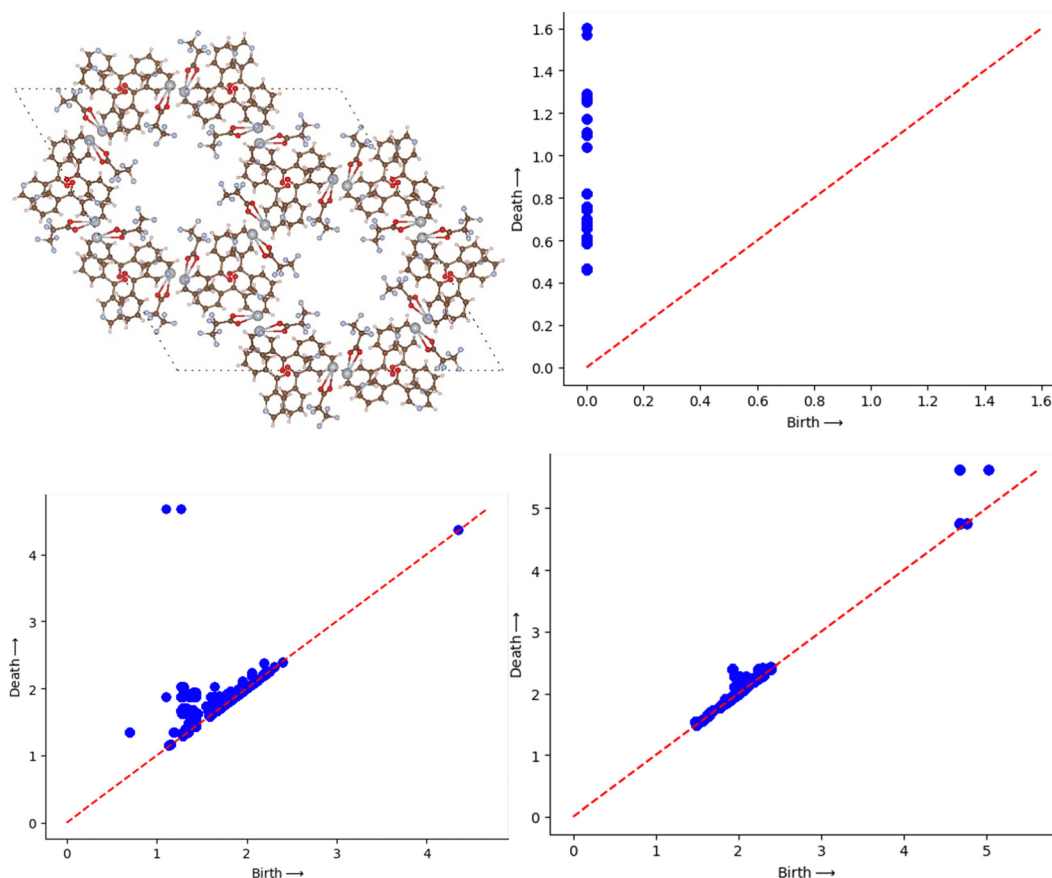


Fig. 2 Structure of FURRUG CoRE MOF and the corresponding 0D, 1D, 2D persistence diagrams.



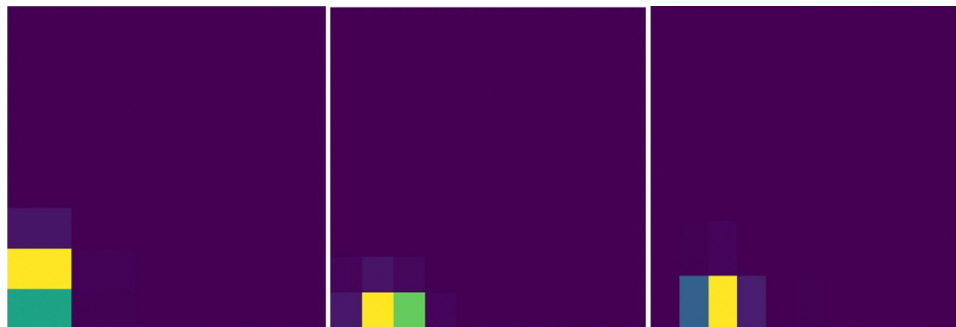


Fig. 3 0D, 1D and 2D persistence images of FURRUG CoRE MOF.

numbers of filters, followed by a global average pooling layer and a fully connected layer for classification. Layers such as convolution, maxpool, fully linked, and softmax are present in the network. It has been used for a wide range of computer vision tasks, including image classification, object detection, and semantic segmentation. ResNet-18 is a relatively shallow architecture compared to some of the larger ResNet variants like

ResNet-50 or ResNet-101. However, it still achieves state-of-the-art performance on many computer vision benchmarks.^{44–46} Fig. 4 represents the layered approach of ResNet-18 model.

In our method we used the ResNet-18 architecture to extract meaningful feature representation from the images, we use half the number of features used at each layer in the vanilla implementation and after Average pooling, we concatenate

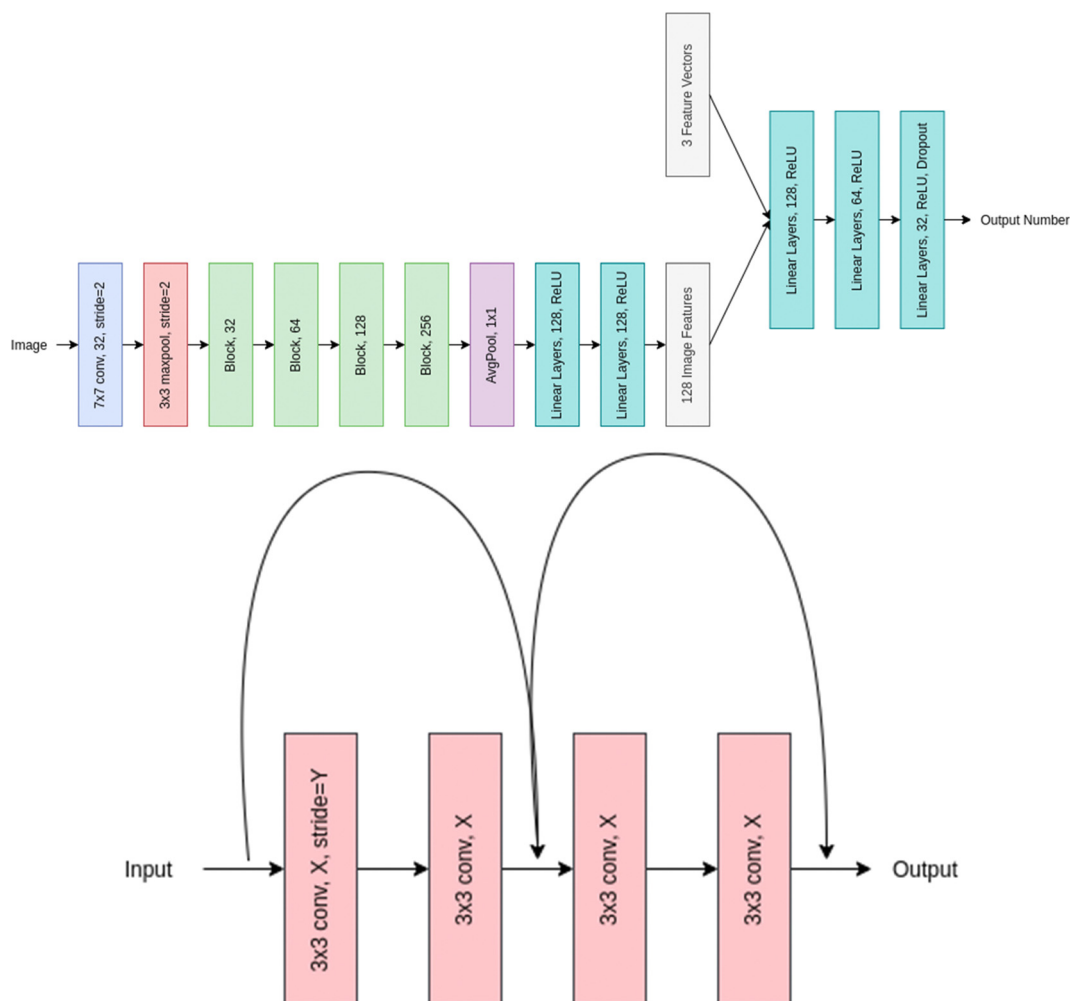


Fig. 4 Overall representation of ResNet-18 model (upper panel) and Block XY of the model where X and Y represent the filter number and stride used respectively.



Table 1 Details of network parameter values used in our ResNet-18 model

Parameter	Value
Initial learning rate	1×10^{-3}
Dropout ratio	0.2
Bias	True
Batch size	32
Number of epochs	200
Activation	ReLU
Number of layers	21
Random state	10
Batch normalization	True
Trainable parameters	3.222017

the image feature vector with our feature vector, and then pass this through multiple linear layers to get our final out. The XY in Block XY in the architecture diagram represents filter numbers and stride used respectively. We use a stride of one for the first block layer and 2 for the rest.

We finally apply the Average pool operation to the output to get a 256-dimensional feature map which is passed through two linear layers with ReLU activation to get 128 feature maps which is further concatenated with the feature vectors and downsized *via* the linear layers to finally output a single number. Table 1 details the obtainable number of trainable parameters and the corresponding convolutional layers.

The ResNet-18 architecture first includes a convolutional layer of filter size 7×7 with a stride of 2 and padding of 3 this layer applies 32 such filters and these outputs are then passed to a batch normalization layer, whose output is again passed through a ReLU activation function followed by a Maxpool layer of kernel size of 3 and stride of 3 with padding of 1. The output obtained after doing the above is then passed through a series of Blocks which have 4 convolutional layers each of different filter numbers. Every block's first convolutional layer will have a stride of 2 other than the first block which has a stride of 1, the filter size for each of the blocks is 3×3 with stride and padding set to 1.

In any deep learning model, the hyperparameters play a crucial role in deciding how well the model performs. It is possible to fine-tune the model to achieve optimal performance by adjusting the hyperparameters, which include the training accuracy, training loss, validation accuracy, validation accuracy, batch size, learning rate, and the number of training epochs. In our model, we have used optimised hyperparameters.

Another important thing is, the model's effectiveness relies heavily on its learning rate. This slow training procedure means that the corresponding network weights won't be modified very quickly once they've been learned. The results of a higher learning rate, on the other hand, are likely to deviate from expectations. The learning rate is determined by optimising and minimising the loss function of the neural network. In the current experimental setup, we assume an initial learning rate of 0.001 and conclude that the models and learning rate have reached saturation after 200 training iterations.

We leverage the ResNet-18 architecture to incorporate the feature vectors along with the images, we first pass the images

alone to the ResNet model and extract final features which have been passed through linear layers to restructure it to the required dimension, this image feature vector is then concatenated with the original feature vectors without any preprocessing to give us our final feature vector which is subsequently passed through a neural network that finally predicts one value. We used mean squared error (MSE) as our loss metric and which is determined by the equation given eqn (1) where y_i , y_i^* and m denote the true values, predicted values and the number of samples in the dataset, respectively. The model is trained end to end using images and feature vectors.

$$\text{MSE} = \frac{1}{m} \sum_{i=1}^m (y_i - y_i^*)^2 \quad (1)$$

3 Results and discussion

3.1 Effect of the training set on ML predictions

For accurate ML-based property prediction, a high-quality training set, consisting of a subset of the data, is necessary. The availability of sufficient data about the complete set is an important feature of a successful training set. In this article, we examine the differences between randomly selecting a training set and selecting it based on diversity with the help of a min-max algorithm. Training sets of 1000 structures are used in both methods. As discussed previously, we chose ResNet-18 technique for our ML model. We used the ResNet-18 model to predict H_2 deliverable capacity, and we did so using two distinct methods for selecting a training set: one that only used structural feature vectors (Structural Descriptor or SD) and another that only used image vectors (Image Descriptor or ID). The accuracy of the predictions for both scenarios is comparable. For the remainder study, we adopt a random selection technique due to the computational expense of creating a training set using a diversity selection algorithm.

3.2 Ability of model to distinguish among different images

We need to know if our model can tell the difference between a persistence image and a random image in order to ensure its accuracy in predicting H_2 deliverable capacity, which is the primary goal of this paper. The comparison between regular vectors and regular vectors with images is the most crucial part, so it's vital to make this distinction. Below we plot the matrix which is acting on the concatenated feature vector:

We used 7 image features, we see that the matrix has relatively high values for only a few parts of the image features and input vector features which signifies that the neural network is focusing only on the important part of the image and feature vector and ignoring the rest. This does not represent overfitting as the weights are not too large. This signifies that the model is not totally ignoring the other features; rather, it is giving them a significantly lower weight. The top left side of the right panel image, which corresponds to the dog image feature vectors, appears very dense in comparison to the original persistence image feature vectors (top left side of the left panel image) when we pass random images like dog images instead



of the actual persistence images. This means that the model is giving every image feature almost equal weight, *i.e.* it is not able to extract meaningful features, it is using all the features possible to bring the loss lower. To make up for the picture feature's increased importance, the model is assigning less weight to the normal feature vectors (top right side of the right panel), resulting in a sparser representation. We also get a higher loss for the dog images than the actual feature image. In order to better highlight the effect, we repeated the trials with 128 image features, and the results are displayed in the bottom panel of Fig. 5.

3.3 Comparison of prediction performances

After being confirmed about the validation of our model, we then proceed for further study. First we train our neural network model to predict the hydrogen deliverable capacity of MOFs in the unit namely, UG at TPS (usable gravimetric hydrogen capacity for the temperature + pressure swing between 100 bar/77 K and 5 bar/160 K in units of weight percent) using only the conventional feature vectors. In order to train the neural network, a five-fold cross-validation method is used. Our model exhibits consistent behaviour throughout all five folds. Initially we have chosen 7 feature vectors of the MOFs as discussed in the earlier section. We have seen a validation loss of around 0.069 using these 7 feature vectors.

Then we can see that pore volume (PV), void fraction (VF), and density (d) are the most relevant features by applying the feature importance values supplied to the individual features by our model. The importance of these features can be rationalized by two factors. First, based on the empirical Chahine rule, the pore volume of a MOF correlates with its excess uptake. Second, pore volume and void fraction are related (since $PV = VF \times d^{-1}$) – MOFs with larger PV have larger VF, and *vice versa*. Nevertheless, VF, PV and d remain the three most important single features for both UV and UG conditions, in that order.

From Fig. 6, it is seen that the 3 features contribute mostly to the target property predictions. The testing loss while utilising 3 feature vectors is 0.059 respectively as shown in Table 2. Now, while using the persistence images as image feature vectors taking into consideration of all 0D, 1D and 2D topological features along with the traditional structural feature vectors, it is seen that test loss decreases to 0.043 value which shows the complementarity of image features along with conventional features. This is shown in Fig. 6. To confirm more about our prediction, we have used same dataset for different condition and the prediction here is the usable hydrogen storage capacity of MOFs at 77 K for the pressure swing between 100 and 5 bar (PS condition). Interestingly here also, using persistence images along with feature vectors we are getting a good

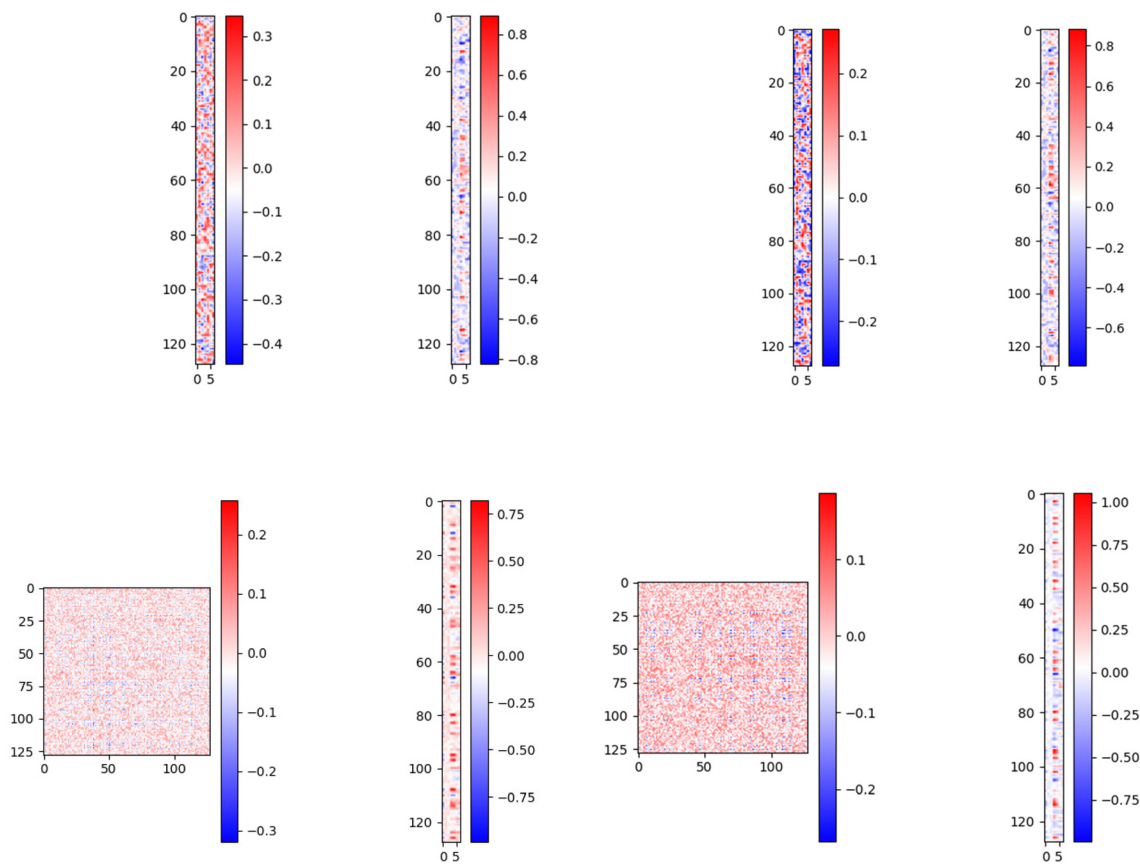


Fig. 5 Matrix plot on concatenated 7 image feature vector taking persistence images (top left panel) and random dog images (top right panel) and 128 image feature vector using persistence image (bottom left panel) and random dog images (bottom right panel).



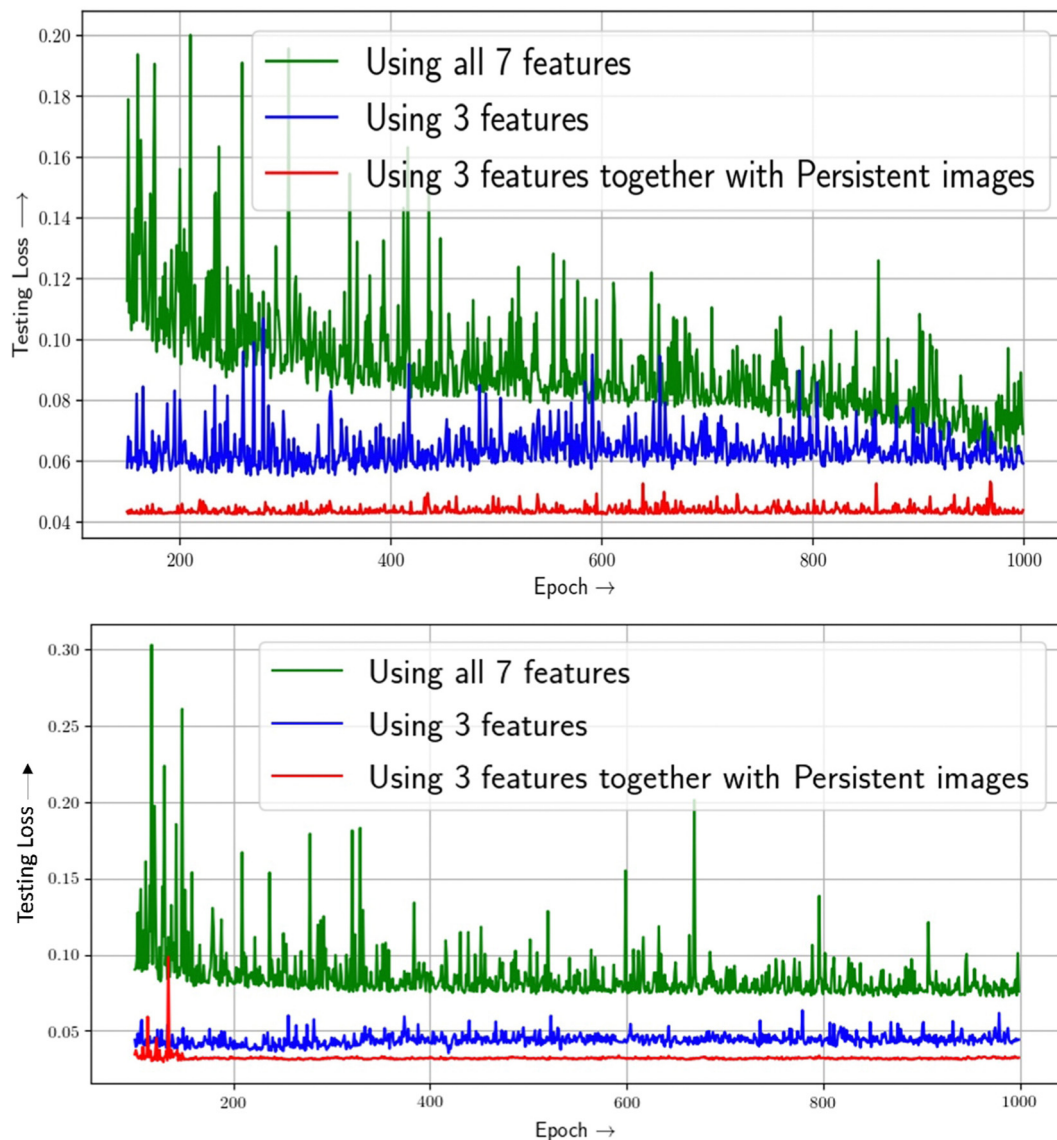


Fig. 6 Multivariate feature importance on predicting H_2 deliverable capacity of MOFs at TPS condition (upper panel) and PS (lower panel) conditions.

Table 2 Performance of ResNet model in predicting H_2 storage capacity in TPS and PS conditions with only features as well as features and topological descriptors

Condition	Method	Test loss
TPS	7-features	0.069
	3-features	0.059
	3-features+ PI	0.043
PS	7-features	0.076
	3-features	0.044
	3-features+ PI	0.033

accuracy than that of only using feature vectors. Table 2 summarizes the performances of feature vectors with that of persistence images. The more improvement using persistence images in PS condition than that of TPS condition is due to the fact that the functional relationships between output capacities

(UG/UV) and input features under PS and TPS conditions are likely different, as was observed in previously reported structure(feature)–property(capacity) relationships.¹⁸

Predicting H_2 storage in MOFs is a well-known challenge, and various prior literatures have attempted to do so using an ML model in an effort to reduce both computational and experimental costs.^{41,42,47} Among these, Ahmed *et al.*⁴¹ conducted an in-depth analysis drawing from several existing literatures to forecast under TPS and PS situations, and they concluded that the extensively randomised tree (ERT) model is effective. They calculated RMSE values of 0.18 and 0.23 (capacity unit) for UG under TPS and PS circumstances. By building an ML model on top of a TDA, we can forecast H_2 storage capacity in MOFs with a reasonable accuracy. The prediction of H_2 storage capacity in MOFs is an area of critical research, and in this work we are able, for the first time, to integrate topological information with the state-of-the-art ResNet model, which is a



Table 3 Properties of best five MOFs according to hydrogen storage capacity

MOF	Density	GSA	VSA	VF	PV	LCD	PLD	UG at TPS	UV at TPS
XAHQAA	0.17	6250.1	1065.2	0.95	5.44	23.04	21.61	19.33	15.72
XAHPUT	0.18	6301.4	1125.9	0.94	5.15	21.83	20.59	18.46	14.93
NIBJAK	0.22	5417.2	1210.4	0.94	4.09	32.0	17.55	16.51	13.2
RAVXOD	0.18	3299.1	590.9	0.88	5.02	71.64	71.5	15.45	12.66
RUTNOK	0.24	6199.7	1493.0	0.9	3.73	24.61	14.65	14.89	12.05

proclaimed advanced successful model in computer vision technologies.

4 Detailed analysis of best five MOFs

Table 3 represents the best five MOFs found through our study with their corresponding features. It is found that MOFs with lower density show higher storage capacity. It is due to the fact that the porosity provides more surface area within the material, which is crucial for hydrogen adsorption.

Other than density, high gravimetric surface area in MOFs is crucial for hydrogen storage, particularly for mobile applications such as vehicles, where minimizing weight is essential for operational efficiency and range. This metric indicates more surface area per unit weight, providing more adsorption sites for hydrogen molecules and thus higher storage capacity. While volumetric surface area, which measures surface area per unit volume, is also important, it's secondary to gravimetric considerations in scenarios where weight has a greater impact on performance than the space occupied by the storage system. Consequently, materials with a high gravimetric surface area are favored for their ability to store a significant amount of hydrogen without adding substantial weight to the energy

storage system. This is also confirmed from our study as evident from the values of GSA and VSA where both of which have higher values for all the best five MOFs. In addition to that it is seen that the best MOFs have high VF which can be explained from chemical intuition. MOFs with a high void fraction are particularly effective for hydrogen storage because the vast empty space translates to a higher surface area for the adsorption of hydrogen molecules. This structural characteristic ensures that there are ample sites for the physical adsorption of hydrogen, optimizing the storage capacity. Additionally, a high void fraction allows for more efficient diffusion of hydrogen molecules throughout the MOF, promoting uniform distribution and accessibility to adsorption sites. The delicate balance of having pores just slightly larger than the hydrogen molecules maximizes the van der Waals forces necessary for adsorption without overly restricting or loosening the hydrogen molecules. Consequently, MOFs with large void fractions are typically superior for hydrogen storage, providing both high capacity and fast kinetics, which are essential for real-world energy applications. MOFs with higher PV show higher hydrogen storage property and this is due to the due to the increased space available for hydrogen adsorption. From crucial inspection of Table 3, it is seen that MOFs with low LCD and high PLD show high hydrogen adsorption capacity. The LCD refers to the size of the largest void space within the MOF structure. A lower LCD indicates smaller, more compact cavities, which can be beneficial for maximizing surface area and adsorption sites within a given volume. Meanwhile, a high PLD, which represents the smallest diameter through which a molecule can pass to access a cavity, ensures that hydrogen molecules can easily enter and fill these cavities. This combination of a compact cavity structure with accessible pores allows for efficient storage of hydrogen. The small cavities increase the surface interaction with hydrogen molecules, boosting adsorption capacity, while the larger pore entrances facilitate easy diffusion of hydrogen

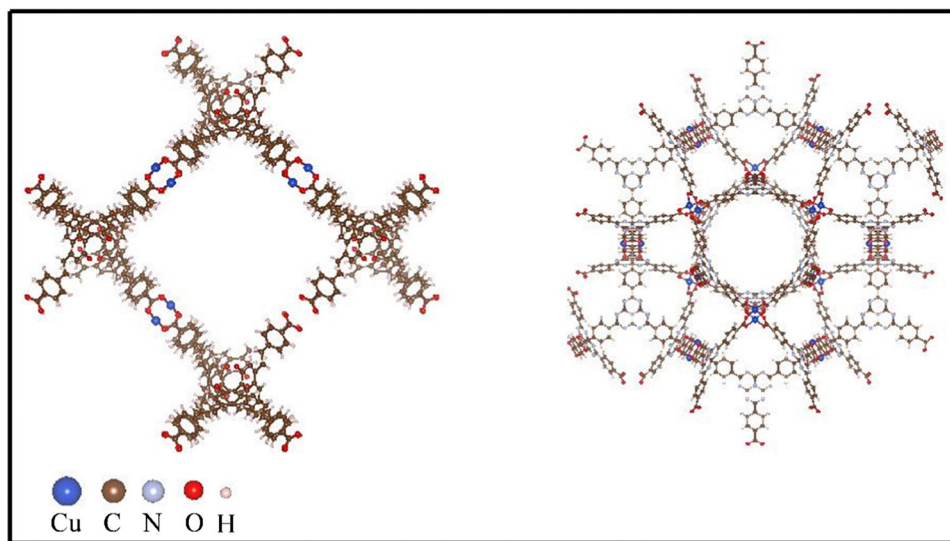


Fig. 7 Structure of two best MOFs in terms of high hydrogen storage capacity. Left panel shows the structure of XAHQAA MOF and right panel corresponds to structure of RUTNOK MOF.



- M. Haranczyk, *et al.*, In Silico Screening of Carbon-Capture Materials, *Nat. Mater.*, 2012, **11**(7), 633–641.
- 18 A. Ahmed, S. Seth, J. Purewal, A. G. Wong-Foy, M. Veenstra, A. J. Matzger and D. J. Siegel, Exceptional Hydrogen Storage Achieved by Screening Nearly Half a Million Metal-Organic Frameworks, *Nat. Commun.*, 2019, **10**(1), 1568.
- 19 J. Behler, Perspective: Machine Learning Potentials for Atomistic Simulations, *J. Chem. Phys.*, 2016, **145**(17), 170901.
- 20 A. Deshwal, C. M. Simon and J. R. Doppa, Bayesian Optimization of Nanoporous Materials, *Mol. Syst. Des. Eng.*, 2021, **6**(12), 1066–1086.
- 21 R. L. Martin, B. Smit and M. Haranczyk, Addressing Challenges of Identifying Geometrically Diverse Sets of Crystalline Porous Materials, *J. Chem. Inf. Model.*, 2012, **52**(2), 308–318.
- 22 R. L. Martin, T. F. Willems, L.-C. Lin, J. Kim, J. A. Swisher, B. Smit and M. Haranczyk, Similarity-Driven Discovery of Zeolite Materials for Adsorption-Based Separations, *ChemPhysChem*, 2012, **13**(16), 3595–3597.
- 23 Y. Lee, S. D. Barthel, P. Dłotko, S. M. Moosavi, K. Hess and B. Smit, Quantifying Similarity of Pore-Geometry in Nanoporous Materials, *Nat. Commun.*, 2017, **8**(1), 1–8.
- 24 P. Y. Lum, G. Singh, A. Lehman, T. Ishkanov, M. Vejdemo-Johansson, M. Alagappan, J. Carlsson and G. Carlsson, Extracting Insights from the Shape of Complex Data Using Topology, *Sci. Rep.*, 2013, **3**(1), 1–8.
- 25 M. Nicolau, A. J. Levine and G. Carlsson, Topology Based Data Analysis Identifies a Subgroup of Breast Cancers with a Unique Mutational Profile and Excellent Survival, *Proc. Natl. Acad. Sci. U. S. A.*, 2011, **108**(17), 7265–7270.
- 26 Y. Schiff, V. Chenthamarakshan, S. C. Hoffman, K. N. Ramamurthy and P. Das, Augmenting Molecular Deep Generative Models with Topological Data Analysis Representations, *ICASSP 2022-2022 IEEE International Conference on Acoustics, Speech and Signal Processing (ICASSP)*, IEEE, 2022, pp. 3783–3787.
- 27 M. Kramár, A. Goulet, L. Kondic and K. Mischaikow, Persistence of Force Networks in Compressed Granular Media, *Phys. Rev. E: Stat., Nonlinear, Soft Matter Phys.*, 2013, **87**(4), 042207.
- 28 Y. Lee, S. D. Barthel, P. Dłotko, S. M. Moosavi, K. Hess and B. Smit, High-Throughput Screening Approach for Nanoporous Materials Genome Using Topological Data Analysis: Application to Zeolites, *J. Chem. Theory Comput.*, 2018, **14**(8), 4427–4437.
- 29 X. Zhang, J. Cui, K. Zhang, J. Wu and Y. Lee, Machine Learning Prediction on Properties of Nanoporous Materials Utilizing Pore Geometry Barcodes, *J. Chem. Inf. Model.*, 2019, **59**(11), 4636–4644.
- 30 A. S. Krishnapriyan, M. Haranczyk and D. Morozov, Topological Descriptors Help Predict Guest Adsorption in Nanoporous Materials, *J. Phys. Chem. C*, 2020, **124**(17), 9360–9368.
- 31 A. S. Krishnapriyan, J. Montoya, M. Haranczyk, J. Hummelshøj and D. Morozov, Machine Learning with Persistent Homology and Chemical Word Embeddings Improves Prediction Accuracy and Interpretability in Metal-Organic Frameworks, *Sci. Rep.*, 2021, **11**(1), 8888.
- 32 D. P. Tabor, L. M. Roch, S. K. Saikin, C. Kreisbeck, D. Sheberla, J. H. Montoya, S. Dwaraknath, M. Aykol, C. Ortiz and H. Tribukait, *et al.*, Accelerating the Discovery of Materials for Clean Energy in the Era of Smart Automation, *Nat. Rev. Mater.*, 2018, **3**(5), 5–20.
- 33 A. Agrawal and A. Choudhary, Perspective: Materials Informatics and Big Data: Realization of the “Fourth Paradigm” of Science in Materials Science, *APL Mater.*, 2016, **4**(5), 053208.
- 34 J. Carrasquilla and R. G. Melko, Machine learning phases of matter, *Nat. Phys.*, 2017, **13**(5), 431–434.
- 35 F. Ramezani, S. Parvez, J. P. Fix, A. Battaglin, S. Whyte, N. J. Borys and B. M. Whitaker, Automatic Detection of Multilayer Hexagonal Boron Nitride in Optical Images Using Deep Learning-Based Computer Vision, *Sci. Rep.*, 2023, **13**(1), 1595.
- 36 K. He, X. Zhang, S. Ren and J. Sun, Deep Residual Learning for Image Recognition, *Proceedings of the IEEE conference on computer vision and pattern recognition*, 2016, pp. 770–778.
- 37 M. Shafiq and Z. Gu, Deep residual learning for image recognition: A survey, *Appl. Sci.*, 2022, **12**(18), 8972.
- 38 J. Liang, *Image classification based on resnet*, Journal of Physics: Conference Series, IOP Publishing, 2020, vol. 1634, p. 012110.
- 39 Z. Hu, J. Tang, Z. Wang, K. Zhang, L. Zhang and Q. Sun, Deep learning for image-based cancer detection and diagnosis- a survey, *Pattern Recognit.*, 2018, **83**, 134–149.
- 40 G. Guo and N. Zhang, A survey on deep learning based face recognition, *Comput. Vision Image Understanding*, 2019, **189**, 102805.
- 41 A. Ahmed and D. J. Siegel, Predicting Hydrogen Storage in MOFs via Machine Learning, *Patterns*, 2021, **2**(7), 100291.
- 42 K. Salehi, M. Rahmani and S. Atashrouz, Machine Learning Assisted Predictions for Hydrogen Storage in Metal-Organic Frameworks, *Int. J. Hydrogen Energy*, 2023, **48**, 33260–33275.
- 43 H. Adams, T. Emerson, M. Kirby, R. Neville, C. Peterson, P. Shipman, S. Chepushtanova, E. Hanson, F. Motta and L. Ziegelmeier, Persistence Images: A Stable Vector Representation of Persistent Homology, *J. Mach. Learn. Res.*, 2017, **18**, 1–35.
- 44 K. M. Black, H. Law, A. Aldoukhi, J. Deng and K. R. Ghani, Deep Learning Computer Vision Algorithm for Detecting Kidney Stone Composition, *BJU Int.*, 2020, **125**(6), 920–924.
- 45 S. Bianco, R. Cadene, L. Celona and P. Napoletano, Benchmark Analysis of Representative Deep Neural Network Architectures, *IEEE Access*, 2018, **6**, 64270–64277.
- 46 A. Z. D. Costa, H. E. H. Figueroa and J. A. Fracarolli, Computer vision based detection of external defects on tomatoes using deep learning, *Biosyst. Eng.*, 2020, **190**, 131–144.
- 47 B. J. Bucior, N. S. Bobbitt, T. Islamoglu, S. Goswami, A. Gopalan, T. Yildirim, O. K. Farha, N. Bagheri and R. Q. Snurr, Energy-Based Descriptors to Rapidly Predict Hydrogen Storage in Metal-Organic Frameworks, *Mol. Syst. Des. Eng.*, 2019, **4**(1), 162–174.

