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## Laser-induced nucleation promotes crystal growth of anhydrous sodium bromide†

Eleanor R. Barber,<sup>a</sup> Martin R. Ward,<sup>b</sup> Andrew D. Ward<sup>c</sup> and Andrew J. Alexander<sup>id</sup>\*<sup>a</sup>

We report on a study of crystal hydrate formation in supersaturated aqueous sodium bromide using different methods to induce nucleation: mechanical shock-induced nucleation (MSIN), nucleation by ultrasound (sonocrystallization), non-photochemical laser-induced nucleation (NPLIN) and laser-trapping nucleation. The most stable crystal form at room temperature is known to be sodium bromide dihydrate (DH) and this form was favoured (>95%) through spontaneous nucleation or mechanical shock. Sonocrystallization favoured DH crystals (74%). Remarkably both laser-induced nucleation methods showed a strong preference (>90%) for anhydrous (AH) crystals. The nucleation mechanisms are discussed with reference to the solution–solid phase diagram. For laser-trapping nucleation, the results are consistent with previous studies showing that nucleation is preceded by formation of a localised volume of increased solute concentration. The common mechanistic feature linking sonocrystallization, MSIN and NPLIN is cavitation. The preference for AH sodium bromide suggests that nanosecond laser pulses produce cavitation events with more thermal energy compared to the other methods. The results demonstrate the value of laser-induced nucleation in controlling crystal hydrate growth and provide new understanding of the nucleation mechanisms.

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## 1. Introduction

The nucleation of crystals from a solution or melt is a fundamental process that is of huge scientific and industrial importance, but the underlying mechanisms remain poorly understood. A subject of ongoing interest, particularly in the pharmaceutical industry, is selective formation of polymorphs, co-crystals or solvates.<sup>1</sup> Different solid forms have different physicochemical properties such as solubility, dissolution rate and stability.<sup>2</sup> Therefore, where a full characterization of such properties is required, *e.g.*, for active pharmaceutical ingredients, obtaining the correct form is crucial. Polymorph control has been achieved traditionally through optimizing the solvent, supersaturation and temperature;<sup>3–7</sup> or by the addition of seed crystals or additives that modify growth at the crystal faces of specific

polymorphs.<sup>8,9</sup> External effects such as an applied electric field, or ultrasonic or optical irradiation can also influence polymorph and solvate formation.<sup>10–15</sup>

Non-photochemical laser-induced nucleation (NPLIN) is a technique in which a new phase is formed by the exposure of a metastable system to pulses of laser light. The phenomenon was first observed in supersaturated solutions of urea,<sup>16</sup> and has since been observed in a diverse range of systems including solutions of simple salts;<sup>17–22</sup> proteins;<sup>23,24</sup> small molecules in various solvents;<sup>13,25–33</sup> and one-component systems.<sup>34,35</sup> NPLIN typically utilises millijoule, unfocused, nanosecond laser pulses in the visible or near-IR region of the electromagnetic spectrum at a wavelength that is not absorbed by the sample. Nucleation of carbon dioxide gas bubbles has also been reported.<sup>36,37</sup>

A particularly remarkable result of NPLIN is the ability to control the polymorph of crystals through the polarization of the laser light used for nucleation.<sup>13,25</sup> This is known as polarization switching. Circularly polarized (CP) light was found to produce the  $\alpha$  polymorph of glycine, which is the form obtained by spontaneous nucleation, while linearly polarized (LP) light produced the  $\gamma$  polymorph, which is the thermodynamically stable form at room temperature. Polarization switching for glycine occurs only within a narrow window of supersaturation and temperature, and it has been found that switching is difficult to reproduce.<sup>27,30,33,38,39</sup>

<sup>a</sup> School of Chemistry, University of Edinburgh, David Brewster Road, Edinburgh, EH9 3JJ, UK. E-mail: andrew.alexander@ed.ac.uk

<sup>b</sup> Strathclyde Institute of Pharmacy & Biomedical Sciences (SIPBS), University of Strathclyde, 161 Cathedral Street, Glasgow, G4 0RE, UK

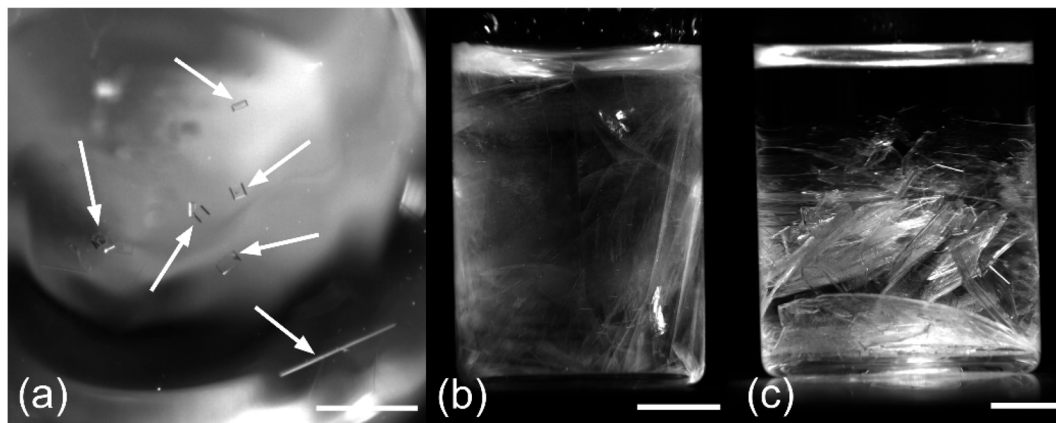
<sup>c</sup> Central Laser Facility, Research Complex at Harwell, STFC Rutherford Appleton Laboratory, Harwell Oxford, Didcot, OX11 0QX, UK

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**Fig. 2** Images of the different crystal morphologies formed from aqueous supersaturated solutions of sodium bromide: (a) small colourless cubic crystals (AH) (scale bar = 2 mm), (b) a translucent mass consisting of flat plates of crystals (DH) that fill the original solution (scale bar = 5 mm), (c) transparent flat plates (DH) leaving a larger fraction of solution at the top (scale bar = 5 mm). Solution conditions:  $C = 11.5 \text{ mol kg}^{-1}$ ,  $T = 21 \text{ }^\circ\text{C}$ ,  $S = 1.29$  (DH) or  $1.02$  (AH).

The AH crystals grew very slowly, and were visible by eye only after 20–30 minutes. However, closer examination of the recorded images showed crystallites falling to the bottom of the vial 10–100 s after the laser pulse. Fig. 3 shows the growth of a typical group of laser-induced AH crystals following exposure to a laser pulse.

### 3.2 Nucleation by mechanical shock and sonocrystallization

Samples that were nucleated deliberately by mechanical shock ( $T = 21 \text{ }^\circ\text{C}$ ) always formed DH crystals. One hit was usually sufficient, although in some cases two or three hits were required. Two different growth behaviours were observed. (1) In some samples, nucleation occurred in localised regions of the solution, often near the solution–air interface (Fig. 4), possibly where liquid had splashed up the walls of the vial. Plate-like crystals grew outwards from these regions until the entire solution was filled with a translucent mass. (2) In other samples, a larger number of nucleation centres were observed throughout the solution (Fig. 5). Smaller plate-like crystals formed and fell to the bottom of the vial: these crystals were smaller and grew more quickly than case (1).

Table 3 shows the number of samples that nucleated after immersion in the ultrasonic bath, compared to the control samples. Only DH crystals were observed, except for ultrasonication at concentrations above saturation for AH ( $11.5 \text{ mol kg}^{-1}$ ) where approximately half of the samples nucleated produced AH crystals.

### 3.3 Laser-trapping nucleation

Only AH crystals were observed in the laser trapping of solutions of NaBr in  $\text{D}_2\text{O}$ . In each of several droplets, a single cubic crystal was observed growing at or near the focal point within 10 s to 4 minutes of exposure to the beam. The growth rate of the crystals could be controlled by adjusting the laser power: at the maximum power (1.2 W) steady growth was maintained; while at lower power (1.0 W) slower growth occurred, and the crystal was less stable in the trap. The crystal could be translated along the interface while trapped as it was pinned by the laser to the interface, but moving the focal volume into the solution resulted in loss of trapping. Trapped crystals were observed to grow out from the focus of the laser, as shown in Fig. 5 (see Video S5 in ESI†). When nucleation was first observed, the crystals produced ( $<1 \text{ }\mu\text{m}$ )

**Table 1** NPLIN of NaBr crystals in the range of temperatures 45–60  $^\circ\text{C}$ . The nominal supersaturation of the solution ( $11.5 \text{ mol kg}^{-1}$ ) with respect to each crystal form is shown (DH = dihydrate; AH = anhydrous). The table shows the number of vials that nucleated after exposure to a single laser pulse (532 nm), in comparison to a control experiment with no laser. Samples that nucleated during the initial hour of cooling prior to exposure were excluded, hence the variation in the total number of samples tested

Temperature ( $T$ )/ $^\circ\text{C}$	Supersaturation ( $S$ )	Control			With laser		
		Total tested	Nucleated		Total tested	Nucleated	
			DH	AH		DH	AH
60	1.004 (AH)	7	0	0	10	0	3
55	1.01 (AH)	10	0	0	10	0	4
50	1.02 (DH)	20	0	0	19	0	8
	1.01 (AH)						
45	1.06 (DH)	10	2	1	5	2	1
	1.01 (AH)						



**Table 2** NPLIN of NaBr crystals at 21 °C. The supersaturation with respect to each crystal form is shown (DH = dihydrate; AH = anhydrous). The table shows the number of vials of solution that nucleated (out of the total tested) after exposure to a single laser pulse (1064 nm). Samples that nucleated spontaneously during the slow cooling overnight, prior to testing, were not counted

Concentration (C)/mol kg <sup>-1</sup>	Supersaturation (S)	Total tested	Nucleated with laser	
			DH	AH
11.5	1.29 (DH)	30	1	21
	1.02 (AH)			
10.9	1.22 (DH)	6	1	0
	0.97 (AH)			
10.0	1.12 (DH)	4	0	0
	0.89 (AH)			

tended to reorientate or spin in the trap due to asymmetry in the shape of the crystal. After some growth (>3 μm) a more-stable trapping configuration was obtained, with the crystal being pinned at an edge or corner, and sometimes jumping from trapping at one corner to another.

## 4. Discussion

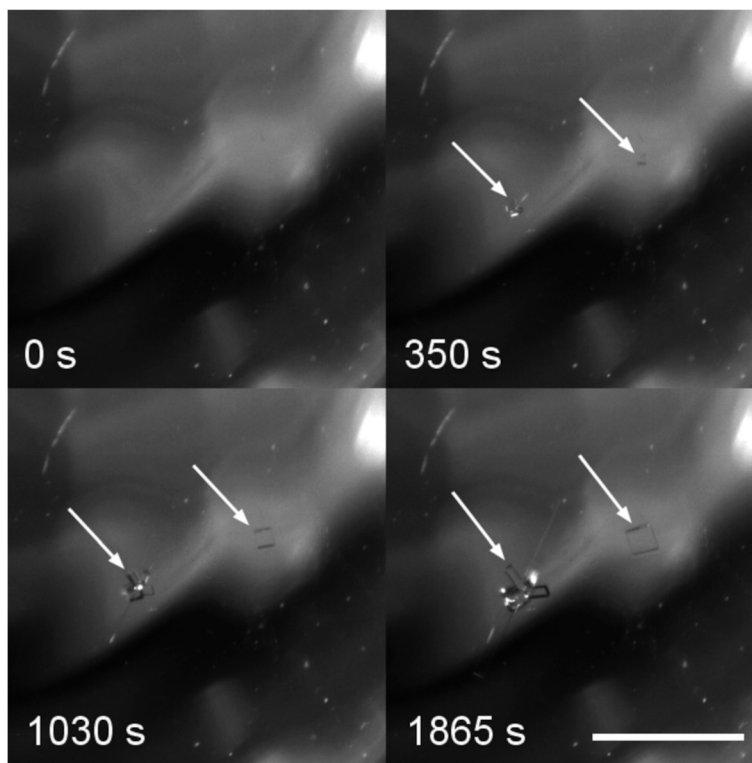
A summary of the observed preferences for hydrate formation using the different methods is given in Table 4. We can group the methods into three categories: (1) strong

preference for DH crystals (spontaneous and mechanical shock); (2) both DH and AH can be formed (sonocrystallization); (3) strong preference for AH crystals (NPLIN, laser trapping). In the remainder of the discussion, we use the results to extract features of the mechanisms involved.

### 4.1 Laser-trapping nucleation

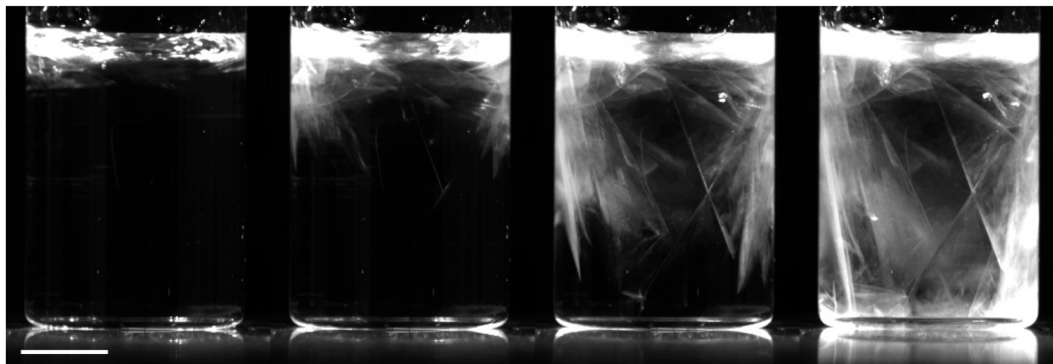
The preference for nucleation of AH *via* laser-trapping is remarkable because the starting solution at  $T = 21$  °C was supersaturated with respect to DH ( $S = 1.08$ ) but undersaturated with respect to AH ( $S = 0.88$ ). The ability to nucleate crystals in undersaturated solutions was first reported by Sugiyama and co-workers for glycine in D<sub>2</sub>O.<sup>54</sup> The CW laser at 1064 nm is expected to cause localised heating due to absorption by the water (vibrational overtones), but the use of D<sub>2</sub>O rather than H<sub>2</sub>O limits this effect to approximately 2 K W<sup>-1</sup> of laser power.<sup>51,52</sup> Therefore, the observed preference cannot be explained simply by elevation of solution above the transition temperature of 47 °C, where AH becomes the preferred solid form in D<sub>2</sub>O (Fig. 1).

Yuyama *et al.* have conducted detailed studies on laser-trapping nucleation of the hydrate system L-phenylalanine (L-Phe) in water.<sup>15,46,51,55–57</sup> Monohydrate (MH) L-Phe is the stable crystal form at room-temperature, and anhydrous L-Phe (AH) is the stable form above 36 °C. For L-Phe in H<sub>2</sub>O a



**Fig. 3** Images of crystal growth of NaBr (AH) from an aqueous supersaturated solution following NPLIN. Solution conditions:  $C = 11.5$  mol kg<sup>-1</sup>,  $T = 21$  °C,  $S = 1.29$  (DH) or 1.02 (AH). The crystals were imaged through the bottom of the vial and are shown between 0 and 31 minutes after the laser pulse (1064 nm). Crystals were first visible approximately 25 s after the laser pulse. Scale bar represents 2 mm.





**Fig. 4** Nucleation by mechanical shock case (1) where nucleation occurred near the solution-air interface. Images of crystal growth of NaBr (DH) from an aqueous supersaturated solution following nucleation by mechanical shock, from left to right: 3.4, 9, 14, and 19 s after nucleation. Solution conditions:  $C = 10.9 \text{ mol kg}^{-1}$ ,  $T = 21 \text{ }^\circ\text{C}$ ,  $S = 1.22$  (DH) or 0.97 (AH). Scale bar represents 5 mm.

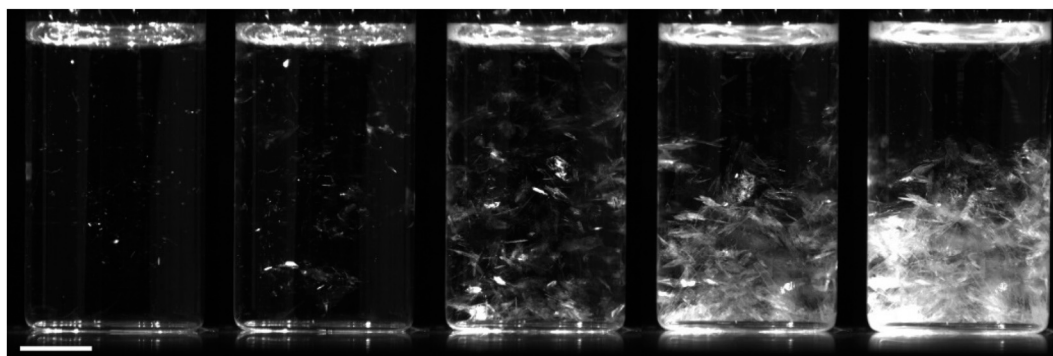
plate-like AH crystal was formed at the focal point, which grew out from its centre in all directions.<sup>46</sup> For L-Phe in  $\text{D}_2\text{O}$ , needle-shaped MH crystals nucleated, but only far away ( $>500 \text{ }\mu\text{m}$ ) from the laser focal volume.<sup>51</sup> It was considered that in  $\text{D}_2\text{O}$ , the interaction between laser and solute at the focal point is much stronger than in  $\text{H}_2\text{O}$ , to the extent that nucleation is hindered there. Nucleation may then occur outside the focal area, where the local structure of the solution favours MH. In the present work, we see growth of a cubic crystal outwards from a trapped corner (Fig. 6), which is likely due to the cubic morphology of AH NaBr *versus* the plate-like morphology of AH L-Phe.

Cheng *et al.* studied laser-trapping nucleation of KCl in  $\text{D}_2\text{O}$ .<sup>52</sup> At low laser powers (0.4 W) the crystal morphology was needle-shaped, but at higher laser powers (1.2 W) a cubic morphology was obtained, similar to the present work. KCl is not known to form hydrates under ambient conditions, and the needle morphology was attributed to growth at high supersaturation. The high concentration was possibly built up over tens of minutes of exposure to the laser prior to nucleation.

Based on previous mechanisms described by Sugiyama and co-workers, we consider that for aqueous NaBr solutions,

the laser acts to build-up concentration of solute around the focal volume.<sup>54</sup> By the time nucleation happens, the local concentration is above the AH solubility line. This can be thought of as an almost vertical transition on the phase diagram, from the bulk concentration (red triangle, Fig. 1) to above the AH solubility curve (dashed curve, Fig. 1). In the case of NaBr in  $\text{D}_2\text{O}$ , this transition would require an increase from  $9.51 \text{ mol kg}^{-1}$  to above  $10.8 \text{ mol kg}^{-1}$ , which is a 14% increase in local concentration. The observation that AH nucleates instead of DH is consistent with Ostwald's rule of stages, which states that the more thermodynamically unstable form will tend to nucleate first. Because the trapping method only creates one localised crystal, there are no DH crystals to promote transition to the more stable form.

We note that  $\text{D}_2\text{O}$  was used as solvent in the laser-trapping experiments only to prevent heating during long-term exposure to the 1064 nm CW trapping beam. A similar experiment could be conducted with NaBr in  $\text{H}_2\text{O}$  using a CW beam at a different wavelength (*e.g.*, 532 nm) where heating of the solvent would be minimal. We expect this would produce results identical to the present work, *i.e.*, nucleation of AH crystals.



**Fig. 5** Nucleation by mechanical shock case (2) where nuclei are initially formed in the bulk of the solution. Images of crystal growth of NaBr (DH) from an aqueous supersaturated solution following nucleation by mechanical shock, from left to right: 0.5, 2.8, 10, 17, and 34 s after nucleation. Solution conditions:  $C = 10.0 \text{ mol kg}^{-1}$ ,  $T = 21 \text{ }^\circ\text{C}$ ,  $S = 1.12$  (DH) or 0.89 (AH). Scale bar represents 5 mm.



**Table 3** Nucleation by sonocrystallization. The nominal supersaturation with respect to each crystal form is shown (DH = dihydrate; AH = anhydrous). The table shows the number of aqueous sodium bromide samples that nucleated after immersion in an ultrasonic bath for approximately one second, with a maximum of five consecutive attempts, at 21 °C

Concentration (C)/mol kg <sup>-1</sup>	Supersaturation (S)	Control		With ultrasound			
		Total tested	Nucleated		Total tested	Nucleated	
			DH	AH		DH	AH
11.5	1.29 (DH)	21	7	0	26	13	10
	1.02 (AH)						
10.9	1.22 (DH)	19	11	0	16	12	0
	0.97 (AH)						
10.0	1.12 (DH)	7	0	0	11	4	0
	0.89 (AH)						

#### 4.2 Pulsed-laser NPLIN

The pulsed-laser NPLIN results show a strong preference for AH (90%). This preference can be rationalised in terms of the particle-heating mechanism for NPLIN.<sup>20,37,40</sup> The mechanism we describe here draws from the experiments of Soare *et al.* and corresponding simulations of Hidman *et al.*<sup>58,59</sup> In their work, crystal nucleation was induced following absorption of a focused laser pulse (~5 kJ cm<sup>-2</sup>) in a supersaturated solution, leading to thermocavitation. By contrast, in the present work the energy density ~1 J cm<sup>-2</sup> is substantially lower, and the absorbing medium is not the solution itself but nanoparticles in solution.

The mechanism is illustrated in Fig. 7. A rapid increase in temperature occurs due to absorption of the laser pulse by a solid nanoparticle (NP). Based on previous findings, the NP is likely to be a trace impurity in the solution, such as iron oxide or carbon.<sup>20,39</sup> The transient high temperature produces a layer of superheated layer (SHL) of liquid around the particle, leading to rapid vaporization and formation of an expanding cavity, *i.e.*, thermocavitation.<sup>60</sup> Vaporization continues from an interfacial layer (IL) at the expanding gas-liquid interface, giving rise to increased local supersaturation (Fig. 7c). The IL is a region of dehydration, which leads to the strong preference for AH while hindering formation of DH.

With reference to the phase diagram (Fig. 1) at 21 °C, at the highest experimental concentration (just above the AH solubility curve) 22/30 samples nucleated, of which 21 were AH. Below this line, only 1/10 samples nucleated, producing DH. This means that NPLIN operates in the region near or above the AH line. We should be cautious and note, however, that we only record events that result in successful growth of a crystal. It is possible that nuclei are formed but redissolve before they have grown sufficiently large to survive for us to observe the outcome.

#### 4.3 Sonocrystallization and mechanical shock

The use of sonocrystallization for nucleation and control of particle size is well established, and acoustic cavitation is considered to be the primary cause.<sup>61</sup> The cavitation that takes place is induced by shear from pressure waves

travelling through the liquid. We believe this causes colder and less-dehydrating nucleation events than NPLIN: at least under ultrasonic bath frequency and power conditions used in the present work. The observed branching between DH and AH neatly represents the position of the initial solution on the phase diagram: at experimental points below the AH solubility line only DH was observed; above this line we observed nucleation of almost equal numbers of AH and DH samples. We note that the exposure time to ultrasound in the present work was fixed at 1 s, in order to focus on primary nucleation. Longer exposure times would modify the ratio of AH to DH, since secondary nucleation will also become important.

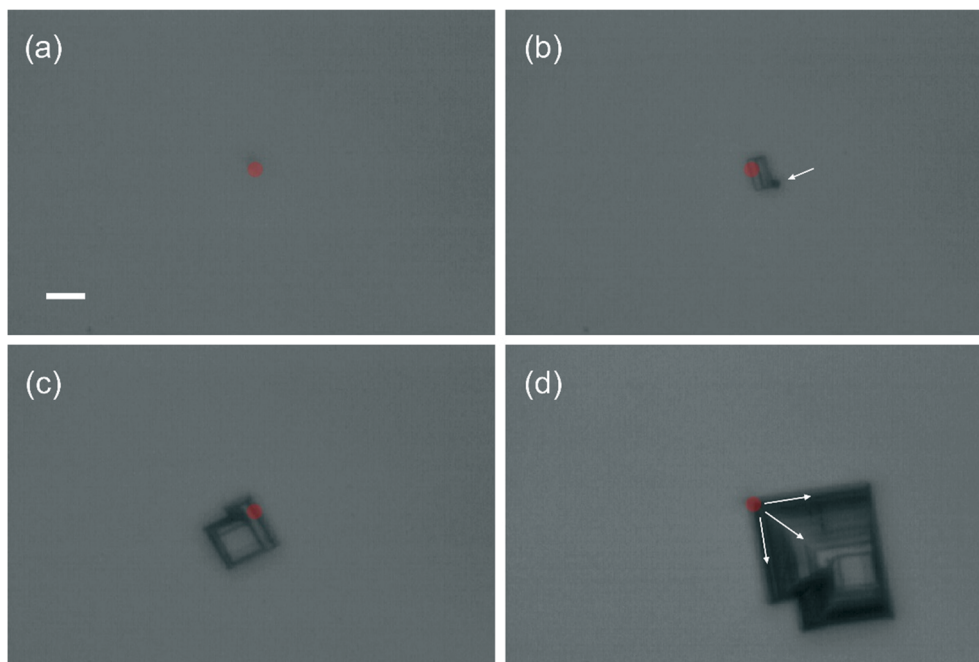
The method we used for MSIN involves striking the base of the vial on a hard surface. A similar method is used in the common prank of beer tapping, where sharp knocking of a carbonated beer bottle releases a foam of bubbles.<sup>62</sup> The rapid acceleration of the vessel causes liquid shear and cavitation. We previously used MSIN to study nucleation of  $\alpha$  and  $\gamma$  polymorphs of glycine.<sup>30</sup> We found that the probability of nucleating  $\gamma$ -glycine crystals increased more sharply as a function of supersaturation using NPLIN compared to MSIN. We interpreted the results by suggesting that NPLIN accessed higher local supersaturations.<sup>30</sup> The results of the present work now suggest that the interface of the expanding cavity in NPLIN is also hotter, more dehydrating, which leads to the preference for AH crystals. Relative to NPLIN and sonocrystallization, we consider shock to cause colder cavitation events, and therefore we observe a stronger preference (100%) for DH.

Finally, we comment on the observations of spontaneous nucleation. Spontaneous, homogeneous

**Table 4** Summary of preferences for NaBr crystal phases using different nucleation methods (DH = dihydrate; AH = anhydrous). For laser trapping, D<sub>2</sub>O was used as solvent; for all the other methods H<sub>2</sub>O was used

Nucleation method	NaBr preferred solid form
Spontaneous	DH (95%)
Mechanical shock	DH (100%)
Sonocrystallization	DH (74%)
NPLIN (pulsed laser)	AH (90%)
Laser trapping (CW laser)	AH (100%)



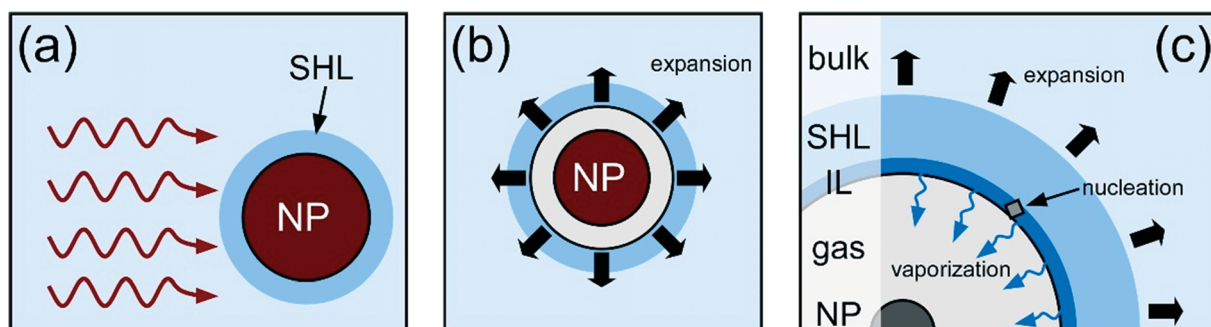


**Fig. 6** Laser-trapping nucleation and crystal growth of NaBr (AH) from an aqueous supersaturated solution in D<sub>2</sub>O. Solution conditions:  $C = 9.51 \text{ mol kg}^{-1}$ ,  $T = 21 \text{ }^\circ\text{C}$ ,  $S = 1.08$  (DH) or  $0.88$  (AH). Laser power was  $1.0 \text{ W}$  at  $1064 \text{ nm}$ . The scale bar represents  $10 \text{ }\mu\text{m}$ . The transparent red dot represents the location of the laser focus in the image plane (focal size not to scale). Images taken at times (a)  $11 \text{ s}$ , (b)  $64 \text{ s}$ , (c)  $78 \text{ s}$ , and (d)  $110 \text{ s}$  after laser was focussed at solution–air interface of droplet. The arrow in (b) shows a small crystallite attached to the main crystal. The arrow in (d) indicates the direction of growth of the crystal, outward from the laser focus, which can be followed by the crystalline steps visible on the crystal. See ESI† (Video S5) for a movie of the nucleation and trapping.

primary nucleation is highly unlikely under normal laboratory conditions.<sup>63</sup> Therefore, heterogeneous mechanisms are more likely to be responsible, *e.g.*, on dust or trace impurities, or cavities. Table 4 shows a strong preference (>95%) for DH for both spontaneous nucleation and mechanical shock. In the present work, movement of the samples was unavoidable, and it's possible that what we call spontaneous nucleation was caused by mechanical shock. It is also possible that spontaneous nucleation is mediated by cavitation events caused by background particle radiation, such as cosmic rays.<sup>60</sup>

## 5. Conclusions

In summary, we have conducted a study of hydrate formation in aqueous sodium bromide. We found that both spontaneous nucleation and nucleation by mechanical shock strongly favoured sodium bromide DH crystals. Using sonocrystallization, the branching between AH and DH crystal forms could be rationalised in terms of the supersaturation with respect to each form, with reference to the solution phase diagram. Laser-trapping nucleation (in D<sub>2</sub>O) showed a complete preference for AH, and it was possible to nucleate solutions that were undersaturated with



**Fig. 7** Schematic diagram of the proposed mechanism for pulsed-laser NPLIN (see section 4.2). (a) A trace nanoparticle (NP) absorbs laser light and the heat transferred to the solution results in a superheated layer (SHL) of liquid. (b) Vaporization in the SHL results in a rapidly expanding vapor cavity. (c) The hot solvent is vaporized from an interfacial layer (IL) at the expanding gas–liquid interface. A crystal nucleus is formed in the IL, which favours AH due to the localised concentration and temperature.





respect to the AH solubility curve. NPLIN with single nanosecond laser pulses was almost entirely ineffective below the AH solubility curve, but gave a very strong preference for AH when supersaturated with respect to this phase. The results suggest that nucleation of crystal hydrates might in general be controlled by choice of method and knowledge of the phase diagram. From the results we have inferred new and useful details of the nucleation mechanisms involved. In particular, the results suggest that NPLIN causes cavitation events where the cavity interface is hotter, and therefore more dehydrating, than in sonocrystallization or nucleation by mechanical shock.

## Data availability

Data employed in this study are available *via* the Edinburgh DataShare repository (DOI: 10.7488/ds/3166x).

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

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