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# KuQuinone-sensitized cobalt oxide nanoparticles for photoelectrocatalytic oxygen evolution with visible light†

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Photocatalytic nanomaterials offer promising solutions for conducting chemical transformations under safe, green and sustainable conditions. In particular, the storage of solar energy into chemical bonds is an appealing but challenging goal in the field of artificial photosynthesis. Using water as the source of electrons and protons through the photodriven water oxidation (WO) reaction is at the core of this endeavour. In this work, we disclose photoactive hybrid nanomaterials designed through a dyadic approach. We exploit  $\text{Co}_3\text{O}_4$  nanoparticles (NPs) covalently functionalized with a fully organic pentacyclic polyquinoid KuQuinone (KuQ) dye, providing a rare example of a noble metal-free photocatalytic dyadic nanomaterial (hereafter denoted as  $\text{KuQ3P}_n\text{@Co}_3\text{O}_4$ ).  $\text{KuQ3P}_n\text{@Co}_3\text{O}_4$  NPs have been characterized by electron microscopy and optical and core-level spectroscopy studies. When cast onto a  $\text{SnO}_2$  photoanode, they are active towards WO upon visible light irradiation (400–580 nm) with a faradaic efficiency for  $\text{O}_2$  evolution of ca. 90%. This work provides a novel contribution to the rational design and mechanistic understanding of hybrid photocatalytic nanomaterials relevant for energy and sustainable synthesis applications.

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## Green foundation

1. Our work advances the field of green chemistry by designing and characterizing a novel hybrid photocatalytic nanomaterial ( $\text{KuQ}_n\text{@Co}_3\text{O}_4$ ) that enables solar-driven water oxidation using sustainable, noble metal-free components. This approach supports the development of artificial photosynthetic systems to store solar energy in chemical bonds, addressing energy sustainability challenges.
2. The  $\text{KuQ3P}_n\text{@Co}_3\text{O}_4$  nanomaterial achieves photoelectrochemical water oxidation under visible light irradiation (400–580 nm) with a faradaic efficiency of ~90%. This highlights its high efficiency as a green catalytic system for solar energy conversion without reliance on precious metals.
3. These findings demonstrate the potential of our dyadic approach for the catalytic activation of small molecules and open the path to greener organic dyes/Earth-abundant metal hybrids. Future research could enhance the sustainability of this work by incorporating naturally derived organic dyes to minimize environmental impact.

## Introduction

Artificial photosynthesis aims at the efficient storage of solar energy into chemical bonds and is considered a primary goal

in the scope of a sustainably powered society. The core target is the utilization of naturally abundant raw materials such as water, carbon dioxide or dinitrogen to produce energy-dense chemical vectors and valuable chemicals.<sup>1</sup> These reductive

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transformations are typically coupled to the oxidation of readily available water to oxygen (WO,  $E_{\text{O}_2/\text{H}_2\text{O}}^\circ = 1.23 \text{ V}$  vs. the reversible hydrogen electrode, RHE; WO is also reported in the literature as the oxygen evolution reaction, OER), providing the reducing equivalents required to feed fuel synthesis.<sup>2,3</sup> A general approach to drive sunlight-induced redox reactions is the integration of a light harvesting unit (LHU) and a water oxidation catalyst (WOC) through chemical interactions.<sup>4–9</sup> This dyadic approach allows the attainment of a controllable LHU/WOC ratio and interaction to promote the desired photoredox events: light absorption by the LHU followed by charge separation to funnel oxidizing equivalents to the WOC, enabling the  $4\text{e}^-/4\text{H}^+$  WO reaction.<sup>10</sup>

Over several decades, the principles of natural photosynthesis have been followed in the design of biomimetic LHU/WOC assemblies, starting with dyads based on covalently bound molecular components.<sup>11–16</sup> However, covalently assembled molecular dyads suffer from heavily demanding synthetic routes that may afford only a limited degree of tuning of the properties of the resulting systems, especially of the LHU/WOC ratio, compared to assemblies produced *via* supramolecular interactions.<sup>17</sup> Furthermore, stability issues may impair the outcome of elegant design work required for the covalent bridging of the dye and catalyst fragments. Notable recent examples of LHU/WOC integration deal with the supramolecular electrostatic assembly of stable cationic perylene bisimide chromophores and a ruthenium polyoxometalate WOC (quantasome),<sup>18,19</sup> providing more convenient access to photoelectrochemical WO on semiconductor slides. A related strategy was followed in the self-assembly of a LHU/WOC layer, driven by hydrophobic interactions or inclusion complex formation on the surface of nano-SnO<sub>2</sub> or TiO<sub>2</sub> photoanodes, respectively. The devices featured a Ru<sup>II</sup> tris-bipyridine

dye and a Ru bipyridine dicarboxylate WOC modified with aliphatic chains<sup>20–22</sup> or  $\beta$ -cyclodextrin.<sup>23</sup>

The literature addressing the design of photocatalytic assemblies also reports the combination of molecular dyes and metal oxide nanoparticles (NPs) as LHU and WOC units, respectively.<sup>24</sup> However somewhat scarcer, such examples are appealing since nanostructured catalysts can be easily prepared and display higher density of active sites, resulting in higher current densities than molecular complexes in operational devices, along with improved stability.<sup>25–31</sup> Seminal studies by Mallouk and co-workers took advantage of Ru (bpy)<sub>3</sub>-type derivatives as LHUs and colloidal hydrated iridium oxide (IrOx) as a WOC. A tailored synthesis method employed Ru<sup>II</sup> polypyridine complexes bearing dicarboxylic acid groups in the periphery of the bipyridine ligands, acting as capping agents in the synthesis of IrOx NPs<sup>32</sup> (Scheme 1A). The advantage of chemically linking the LHU and WOC moieties was supported by an enhancement of *ca.* two orders of magnitude of the electron transfer (ET) rate from IrOx to the photochemically generated oxidized Ru<sup>III</sup> dye in colloidal suspensions with respect to the “unbound” [Ru(bpy)<sub>3</sub>]<sup>2+</sup> dye.<sup>33</sup> Based on these results, heteroleptic Ru<sup>II</sup> polypyridine complexes were devised, bearing both malonate stabilizing groups during the IrOx NP synthesis and a phosphonate anchor for post-synthetic connection to TiO<sub>2</sub> films.<sup>34–36</sup> Such a strategy allowed the integration of these materials within a dye-sensitized photoelectrochemical cell (DSPEC)<sup>37</sup> photoanode (Scheme 1B).<sup>38</sup> Later developments exploited IrOx and RuO<sub>2</sub> WOCs functionalized on their surface by either free-base or metalated (with either Zn or Pd) porphyrin dyes.<sup>39–42</sup> Despite the encouraging results of these preliminary studies, these systems based on organic dyes were not further investigated in photoinduced WO.

A key step in the development of photoactive hybrid nanomaterials for WO was published by some of us.<sup>43</sup> We reported the synthesis of cobalt spinel-structured NPs sensitized *via* surface decoration with phosphonated analogues of [Ru(bpy)<sub>3</sub>]<sup>2+</sup> (**RuPS@Co<sub>3</sub>O<sub>4</sub>**) (Scheme 1C). These NPs were studied under visible light irradiation, catalyzing WO through light absorption by the ruthenium complex in the presence of persulfate anions as electron acceptors. The direct electronic interaction between the dye and the catalyst was clearly demonstrated, alongside the superior catalytic performance of the dyadic **RuPS@Co<sub>3</sub>O<sub>4</sub>** NPs compared to the “unbound” system, *i.e.*, co-suspended **RuPS** and Co<sub>3</sub>O<sub>4</sub> NPs, with photocatalytic WO being relevant only in the former, dyadic case.

Finally, in 2021, Amiens and co-workers reported an analogous strategy applied to core-shell iron-iron oxide (Fe-FeOx) NPs stabilized by a phosphonate-bound Ru phenanthroline dye<sup>44</sup> (Scheme 1D). These composites were employed under photoelectrochemical conditions upon deposition on fluorine-doped tin oxide (FTO). They displayed photocurrent activity associated with O<sub>2</sub> evolution, thereby reinforcing the applicability of nanostructured WOC-molecular LHU combinations to both colloidal and DSPEC artificial photosynthetic reaction architectures.



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**Scheme 1** Photo(electro)catalytic dyadic systems for water oxidation (WO) based on transition metal oxide NPs functionalized with Ru-based molecular dyes (LHU) (top, A–D) and the KuQ3P@Co<sub>3</sub>O<sub>4</sub> hybrid nanomaterials presented in this work (bottom).

These last two works were the first examples of photo-catalytic dyadic materials for WO exploiting a first row-based metal oxide WOC, although still taking advantage of a LHU based on a scarce transition metal like ruthenium. This drawback, together with the need to exploit harness photons beyond 500 nm, prompted us to investigate the use of highly absorbing organic dyes for the design of LHU/WOC composites.<sup>45–47</sup>

Nanostructured Co<sub>3</sub>O<sub>4</sub> is a promising electrocatalyst, able to oxidize water to O<sub>2</sub> with overpotentials as low as 350 mV and operating in a broad pH range, including acidic conditions.<sup>48–52</sup> Thus, Co<sub>3</sub>O<sub>4</sub> represents a viable alternative to precious Ir and Ru-based WOCs. Furthermore, the solid mechanistic framework on Co<sub>3</sub>O<sub>4</sub> as the WOC built by means of spectroscopic, spectroelectrochemical and computational investigations provides a foundation for system design and optimization.<sup>53–60</sup> Herein, we employed Co<sub>3</sub>O<sub>4</sub> NPs synthesized *via* the organometallic approach,<sup>43,61,62</sup> *i.e.*, the controlled decomposition of an organometallic Co compound under H<sub>2</sub> pressure in the presence of a stabilizing ligand followed by air exposure. This routine allows control over particle size, shape, and surface composition with high reproducibility, affording well-defined NPs whose catalytic properties can be fine-tuned

through post-synthetic modifications. Subsequently, we undertook the synthesis of photocatalytic dye–catalyst hybrid nanomaterials incorporating several chromophore units bonded to individual catalyst NPs, designed with the aim of providing a controllable catalyst/dye ratio and a fast ET pathway between the two components.

We employed organic dyes denoted as KuQuinones (**KuQs**),<sup>63,64</sup> since they are well-established active units in photo(electro)catalytic systems that are suited for performing photoinduced WO.<sup>65–70</sup> Indeed, they manage proton-coupled electron-transfer (PCET) events both in their ground and excited states, which are highly valuable in the scope of multi-electron, multi-proton photoinduced reactions for artificial photosynthesis applications.<sup>63–70</sup> Incidentally, the structural quinoid motif is privileged within the field of functional mimetics of natural photosynthetic systems, employing quinones as photoinduced ET relays. Furthermore, **KuQ** dyes can partake in keto–enol tautomerism and enol–enolate equilibria at the level of the pentacyclic core depending on their protonation state, with the enol and enolate forms displaying different light harvesting features, both exhibiting intense absorption in the visible region up to 600 nm<sup>71,72</sup> (*vide infra*). Their singlet excited states (<sup>1</sup>**KuQ**) are highly oxidizing<sup>67</sup> (>2.6 V *vs.* RHE),



and they can inject electrons from their singly reduced form ( $E_{\text{KuQ}^{\bullet-}/\text{KuQ}}^{\circ} \approx 0.3 \text{ V vs. RHE}$ ) into the conduction band of mesoporous  $\text{SnO}_2$  (conduction band edge potential  $E_{\text{CB}} = 0.50 \text{ V vs. RHE}$ ). Lastly, their synthetic space allows for the introduction of side chains that do not impact the chromophore core, while imparting different acid-base, solubility, and anchoring properties. These features contribute to the appeal of **KuQ** dyes as cutting-edge molecular LHUs for WO applications. Based on these premises, a **KuQ** dye with a pendant phosphonate moiety, 1-[3-(dihydroxyphosphonyl)propyl]KuQuinone (**KuQ3P**),<sup>68,73</sup> was used, exploiting their side chains to act as ligands to the surface of the pre-synthesized  $\text{Co}_3\text{O}_4$  NPs *via* a highly affine and hydrolytically stable anchor.

The resulting noble metal-free **KuQ3P**@ $\text{Co}_3\text{O}_4$  hybrid nanomaterials operate in photoelectrochemical WO with visible light (400–580 nm), and the covalent anchoring of the two components is crucial in achieving a high faradaic efficiency for WO. Spectroscopic evidence confirms the photoelectrochemical formation of Co-oxo/hydroxo active sites.

## Results and discussion

### Synthesis and characterization of **KuQ3P**<sub>*n*</sub>@ $\text{Co}_3\text{O}_4$

The synthesis of spherical cobalt(II,III) oxide NPs was carried out *via* a two-step method (Scheme 2) following adaptation of a previous report.<sup>43,61,62</sup> The first step involved the decomposition of (cyclooctadienyl)(1,5-cyclooctadiene)cobalt(I),  $[\text{Co}(\eta^3\text{-C}_8\text{H}_{13})(\eta^4\text{-C}_8\text{H}_{12})]$  under  $\text{H}_2$  at 3 bars in 1-heptanol (heptOH), which acted both as the solvent and stabilizing ligand. This yielded spherical  $\text{Co}^0$  NPs with a diameter of  $2.9 \pm 0.6 \text{ nm}$ , stabilized by heptOH ( $\text{Co}^{\text{heptOH}}$ ). Subsequently, air diffusion allowed for a controlled oxidation process of the isolated NPs to generate  $\text{Co}_3\text{O}_4^{\text{heptOH}}$ , with no substantial change in the particle size ( $3.2 \pm 0.6 \text{ nm}$ ) and shape. High-resolution transmission electron microscopy (HR-TEM) images of  $\text{Co}^{\text{heptOH}}$  and  $\text{Co}_3\text{O}_4^{\text{heptOH}}$  NPs are provided in Fig. S1†. Considering spherical NPs with a *ca.* 3.0 nm diameter, it was estimated that 39% of total  $\text{Co}_3\text{O}_4$  units are surface-exposed (see the ESI, Section 1.4† for details).<sup>‡74</sup>

The synthesis protocol for producing **KuQ3P**-sensitized NPs (Scheme 2) consisted of stirring  $\text{Co}_3\text{O}_4^{\text{heptOH}}$  NPs in the presence of different amounts of **KuQ3P** (0.1 and 0.2 equivalents of **KuQ3P** with respect to  $\text{Co}_3\text{O}_4^{\text{heptOH}}$ ) for 4 days in a heptOH/MeOH/ $\text{H}_2\text{O}$  mixture, followed by washing and drying (see the Methods section for further details). The phosphonate anchoring group was selected based on a previous report by some of us,<sup>43</sup> which showed a better stability of hybrids with this group compared to the carboxylate one. The complete discolouration of the reaction mixtures (Fig. S2†) and the elemental composition of the resulting products afforded by inductively coupled plasma optical emission spectroscopy (ICP-OES) and ICP coupled to mass spectrometry (ICP-MS, see the ESI and

Tables S1 and S2†) were consistent with the quantitative chemisorption of the dye on the NP surface for both materials. The resulting dye-sensitized NPs are hereafter denoted as **KuQ3P**<sub>*n*</sub>@ $\text{Co}_3\text{O}_4$  (*n* = 0.1 and 0.2, respectively). Based on the fraction of surface  $\text{Co}_3\text{O}_4$  sites, this results in a surface ratio between **KuQ3P** and  $\text{Co}_3\text{O}_4$  units of 0.25 : 1 and 0.5 : 1 for **KuQ3P**<sub>0.1</sub>@ $\text{Co}_3\text{O}_4$  and **KuQ3P**<sub>0.2</sub>@ $\text{Co}_3\text{O}_4$ , respectively (ESI, Section 1.4†).

The dyads were then characterized by means of TEM, X-ray absorption spectroscopy (XAS), X-ray photoelectron spectroscopy (XPS), resonance Raman spectroscopy and attenuated internal reflection Fourier transform infrared spectroscopy (ATR-FTIR) (Fig. 1 and Fig. S3–S16†).

TEM analysis of the **KuQ3P**<sub>*n*</sub>@ $\text{Co}_3\text{O}_4$  NPs revealed no significant changes in the mean particle size with respect to the initial  $\text{Co}_3\text{O}_4^{\text{heptOH}}$  NPs, and a certain degree of particle aggregation resulting from **KuQ3P** functionalization (Fig. S3†). HR-TEM images (Fig. 1A and Fig. S4 and S5†) and electron diffraction patterns (Fig. S6†) of **KuQ3P**<sub>0.1</sub>@ $\text{Co}_3\text{O}_4$  evidenced interplanar distances of 0.249, 0.211, 0.148 and 0.145 nm, with values in agreement with those of the (111), (004), (044) and (044) planes of  $\text{Co}_3\text{O}_4$ , respectively, despite a certain degree of surface amorphization. Additionally, a punctiform envelope surrounding the NPs was detected by High Angle Annular Dark Field-Scanning Transmission Electron Microscopy (STEM-HAADF) (Fig. 1B and Fig. S3†). EDX (energy-dispersive X-ray spectroscopy, Fig. S4 and S5†) analyses indicated the presence of Co in this envelope, its formation being observed upon treatment of  $\text{Co}_3\text{O}_4^{\text{heptOH}}$  NPs with the solvent mixture used for the dyad preparation (Scheme 2), irrespective of the presence of **KuQ3P** (see Fig. S7†).

The electronic and structural properties of the **KuQ3P**<sub>*n*</sub>@ $\text{Co}_3\text{O}_4$  dyads were investigated by XAS, a sensitive probe of the oxidation state, geometry and coordination environment of the absorbing metal center. The X-ray absorption near-edge structure (XANES) of the initial  $\text{Co}_3\text{O}_4^{\text{heptOH}}$  NPs aligns with a spinel structure, containing a mixture of  $\text{Co}^{\text{II}}$  ( $T_d$  sites) and  $\text{Co}^{\text{III}}$  ( $O_h$  sites) in a 1 : 2 ratio, exhibiting an intense 1s-to-3d pre-edge feature and a shifted edge position with respect to the purely  $O_h$   $\text{Co}^{\text{II}}$  references  $\text{CoO}$  and  $\text{Co}(\text{OH})_2$ , consistent with the expected  $\text{Co}^{\text{II}}/\text{Co}^{\text{III}}$  ratio (Fig. 1D). This was confirmed by the extended X-ray absorption fine structure (EXAFS) region of the spectra, which provided additional information about the coordination environment of the Co absorber. The *R*-space spectrum shown in Fig. 1E is dominated by contribution from Co–O scattering vectors at 1.92 Å, and two additional intense peaks at 2.86 and 3.36 Å, attributed to remote Co–Co[ $O_h$ ] and Co–Co[ $T_d$ ] distances, respectively (Fig. S8, S10, and Tables S3, S5†). The bond lengths obtained by EXAFS fitting analysis are in the range of the expected values for  $\text{Co}_3\text{O}_4$  X-ray structures and related EXAFS data in the literature.<sup>75</sup>

Although  $\text{Co}_3\text{O}_4^{\text{heptOH}}$  NPs display a textbook spectrum for a cobalt spinel structure, significant changes in both XANES and EXAFS regions have been observed for their functionalized **KuQ3P**<sub>0.1</sub>@ $\text{Co}_3\text{O}_4$  (Fig. 1D and S8, 9†) and **KuQ3P**<sub>0.2</sub>@ $\text{Co}_3\text{O}_4$

‡ Comparable results were obtained by adopting the approach reported in ref. 74, as described in the ESI, Section 1.4.†





**Scheme 2** Synthesis of KuQ3P<sub>n</sub>@Co<sub>3</sub>O<sub>4</sub> dyads via (1) the organometallic approach, (2) controlled air oxidation of Co<sup>heptOH</sup> NPs and (3) Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup> NP sensitization with KuQ3P.



**Fig. 1** (A) HR-TEM and (B) STEM-HAADF images of KuQ3P<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub> NPs with (C) their corresponding size histogram. (D) Normalized Co K-edge XANES spectra of Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup> NPs (black line), KuQ3P<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub> NPs (blue line), CoO (orange line) and Co(OH)<sub>2</sub> (dark red line) references. Inset: zoom of the pre-edge region. (E) Fourier transforms of the *k*<sup>3</sup>-weighted Co EXAFS spectra for Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup> NPs, KuQ3P<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub> NPs, CoO and Co(OH)<sub>2</sub>, same colour code as in (D). Inset: corresponding *k*<sup>3</sup>-weighted Co EXAFS spectra. (F) Resonance Raman spectra of Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup> NPs (black line), KuQ3P<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub> NPs (blue line) and KuQ3P dye (magenta line). Spectra were recorded on powder samples.

(Fig. S8†) counterparts. The edge and white-line features are shifted to lower energies, indicating a slight reduction of the metal center. Additionally, the quadrupole-allowed but dipole-forbidden 1s-to-3d transitions of the pre-edge are characterized by a clear intensity decrease, suggesting the presence of more centrosymmetric O<sub>h</sub> cobalt sites.<sup>75–77</sup> On the other hand, the XANES spectrum is clearly different from the spectra of CoO or Co(OH)<sub>2</sub>. The absence of CoO/Co(OH)<sub>2</sub> was further reinforced by XPS (Fig. S12†) and resonance Raman spectroscopy measurements (Fig. 1F), *vide infra*.

The *R*-space EXAFS spectra showed a clear amorphization of the Co<sub>3</sub>O<sub>4</sub> NPs, as reflected by the loss of intensity in the EXAFS features relative to the pristine Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup> NPs (Fig. S11†, right). Further fitting of the KuQ3P<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub> EXAFS spectrum to a simple spinel model showed a decrease of the Co–O and Co–Co distances relative to a spinel structure to values of 1.90 and 2.82 Å, respectively (Fig. S10†). This

trend is in good accordance with the HR-TEM images of KuQ3P<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub>, which suggest partial surface fragmentation of Co<sub>3</sub>O<sub>4</sub>, leading to the formation of new single-atomic Co species ascribed to Co<sup>II</sup> and Co<sup>III</sup> sites in accordance with XAS data. The influence of KuQ3P on the evolution of the initial Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup> NPs into a more amorphous material was ruled out, as a similar transformation of both XANES and EXAFS regions was also observed in the absence of KuQ3P (Fig. S11†), in accordance with the HR-TEM data, *vide supra* (Fig. S7†).

Therefore, we can hypothesize that under the conditions applied for the post-functionalization of their surface, the pristine Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup> NPs evolved into a complex mixture of Co<sub>3</sub>O<sub>4</sub> and single/multi-atomic octahedral Co<sup>II</sup> and Co<sup>III</sup> sites, without the direct involvement of KuQ3P in the transformation. Single/multi-atomic octahedral Co<sup>II</sup> and Co<sup>III</sup> species have been reported in the literature, and present short Co–O



and Co–Co distances (1.89 and 2.82 Å, respectively) that could explain the decrease of the overall Co–O and Co–Co distances observed in FT-EXAFS.<sup>78–82</sup> A pre-edge fitting of the **KuQ3P**<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub> feature could be performed to roughly estimate the loss of the Co<sub>3</sub>O<sub>4</sub> *T*<sub>d</sub> sites by combining the spectra of Co<sub>3</sub>O<sub>4</sub> and [Co<sup>II</sup>(MeOH)<sub>6</sub>]<sup>2+</sup> as a purely octahedrally coordinated Co<sup>II</sup> model in a 6 : 4 ratio, suggesting that *ca.* 40% of the Co<sub>3</sub>O<sub>4</sub> surface atoms have been fragmented forming the single/multi-atomic species (Fig. S9 and S10†).

Resonance Raman spectroscopy was employed to probe the **KuQ3P**–Co<sub>3</sub>O<sub>4</sub> surface interaction; spectra recorded for Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup>, **KuQ3P**, and **KuQ3P**<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub> are reported in Fig. 1F (additional spectroscopic data are presented in the ESI, Fig. S13–S15†). The **KuQ3P** signals collected in the 500–1700 cm<sup>−1</sup> region are mostly ascribable to the characteristic vibrational signature of the **KuQ** chromophore core motions based on comparison of **KuQ3P** to other dyes in the **KuQ** category (specifically 1-(3-carboxylpropyl)KuQuinone, **KuQ3C**, and 1-hexylKuQuinone **KuQ-Hex**, Fig. S14†). The spectrum of Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup> NPs (black line in Fig. 1F) displays the expected bands at 485, 524 and 691 cm<sup>−1</sup> (blue line in Fig. 1F), ascribed to the E<sub>g</sub>, F<sub>2g</sub> and A<sub>1g</sub> transitions in Co<sub>3</sub>O<sub>4</sub>.<sup>48,60,83</sup> The spectra of **KuQ3P**<sub>*n*</sub>@Co<sub>3</sub>O<sub>4</sub> (blue line in Fig. 1F for **KuQ3P**<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub>, while the spectrum of **KuQ3P**<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub> is reported in Fig. S13†) show an overlap of the signatures of the Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup> NPs and of the **KuQ3P** dye (magenta line in Fig. 1F), with two relevant features: (i) in **KuQ3P**<sub>*n*</sub>@Co<sub>3</sub>O<sub>4</sub>, the A<sub>1g</sub> band peaking at 680 cm<sup>−1</sup> is slightly redshifted with respect to Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup>, where it is observed at 691 cm<sup>−1</sup>. Such a redshift may be attributed to crystal disorder induced by the functionalization step<sup>83</sup> and (ii) the signals of **KuQ3P** are retained in **KuQ3P**<sub>*n*</sub>@Co<sub>3</sub>O<sub>4</sub>, except for the band at 934 cm<sup>−1</sup>; the latter is attributed to a vibration of the phosphonate group,<sup>84–87</sup> and thus supports the involvement of phosphonate in the covalent bonding to the NPs. Lastly, the resonance Raman spectrum of free **KuQ3P** is characterized by the onset of the expected intense emission in the 2000–4000 cm<sup>−1</sup> range, whereas the spectra are instead strongly quenched in **KuQ3P**<sub>*n*</sub>@Co<sub>3</sub>O<sub>4</sub> (Fig. S15†), suggesting that a photoinduced event is occurring in the latter (*vide infra*).

A comparison of the ATR-FTIR spectra recorded for Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup>, **KuQ3P** and **KuQ3P**<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub> (as a representative of the hybrid nanomaterials) is reported in the ESI (Fig. S16†). These data correlate well with the information provided by resonance Raman spectroscopy. The vibrations of **KuQ3P** observed in the “fingerprint” region were consistently identified also in **KuQ3P**<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub>. Furthermore, the motion of the phosphonic acid group at 934 cm<sup>−1</sup> for the free **KuQ3P** is absent in the spectrum of **KuQ3P**<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub>, further supporting the involvement of phosphonate in surface binding to the NPs. ATR-FTIR spectroscopy, not limited by the insurgence of **KuQ3P** emission, allowed us to characterize the NPs in the region above 2500 cm<sup>−1</sup>. Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup> was found to display characteristic peaks at 2925 and 2855 cm<sup>−1</sup>, attributable to the C–H stretching of the alkyl chain of heptOH; likewise, the broad feature at 3320 cm<sup>−1</sup> is ascribed to the O–H stretching

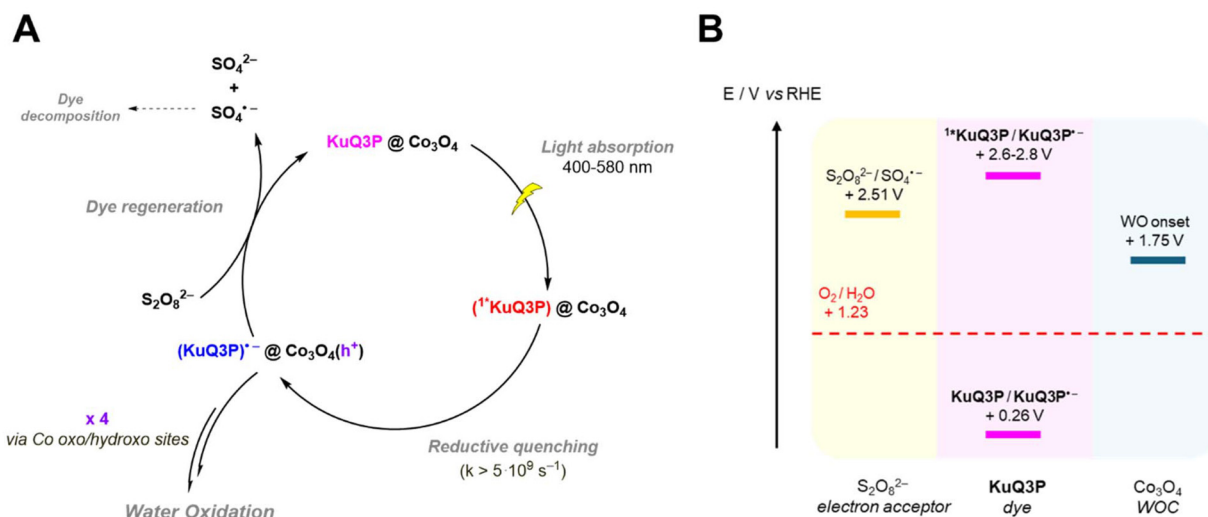
of the heptOH alcohol moiety, consistent with a previous report.<sup>43</sup> Such spectroscopic signatures were found significantly abated in **KuQ3P**<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub>, thus supporting the expected, nearly quantitative ligand substitution of heptOH by **KuQ3P** during Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup> post-functionalization, also in coherence with the ICP-OES results.

Once the successful synthesis of the **KuQ3P**<sub>*n*</sub>@Co<sub>3</sub>O<sub>4</sub> hybrid nanomaterials was confirmed, their photophysical properties were evaluated. In aqueous solution, the **KuQ3P** dye shows a broad fluorescence emission band from the singlet excited state <sup>1</sup>\***KuQ3P**, centred at 620 nm and characterized by a lifetime of 0.51 ns. The fluorescence was strongly abated in **KuQ3P**<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub> with respect to that of the **KuQ3P** dye, suggesting a fast quenching of <sup>1</sup>\***KuQ3P** by the Co<sub>3</sub>O<sub>4</sub> NPs (Fig. S17†).<sup>88</sup> Time-resolved luminescence analysis also suggested ultrafast kinetics (Fig. S17†), with a time-constant  $\tau$  < 0.2 ns, *i.e.*, below the instrumental limit of the time-correlated single photon counting (TCSPC) apparatus, corresponding to a quenching constant lower limit of  $k > 5 \times 10^9$  s<sup>−1</sup>. Considering the high oxidizing power of the <sup>1</sup>\***KuQ3P** excited state ( $E_{1^* \text{KuQ3P}/\text{KuQ3P}^-} = 2.62\text{--}2.79$  V *vs.* RHE), estimated from the combination of the redox potential of the ground state and from the  $E_{0-0}$  of <sup>1</sup>\***KuQ3P** (see the ESI and Fig. S18 and S19†), and considering previous investigations on **KuQ** dyes under homogeneous conditions<sup>70</sup> and at the surface of photoelectrodes,<sup>67</sup> a plausible hypothesis for the observed emission quenching in **KuQ3P**<sub>*n*</sub>@Co<sub>3</sub>O<sub>4</sub> is ET from Co<sub>3</sub>O<sub>4</sub> to the photo-excited **KuQ3P**, leading to oxidized Co<sub>3</sub>O<sub>4</sub>, Co<sub>3</sub>O<sub>4</sub>(h<sup>+</sup>), and **KuQ3P**<sup>•−</sup> (reductive quenching of the dye, as depicted in Scheme 3).<sup>67</sup> This proposed preferential photoinduced ET pathway is exergonic by *ca.* 1 eV, as estimated from the difference between the potential of <sup>1</sup>\***KuQ**/**KuQ**<sup>•−</sup> and the potential threshold to access the catalytic WO regime by Co<sub>3</sub>O<sub>4</sub> (*ca.* 1.75 V *vs.* RHE, as estimated by cyclic voltammetry, CV, see Fig. S20†). The saturated chain linker of the phosphonate anchoring group is not expected to mediate the photoinduced ET.

### Photochemical water oxidation and evolution of Co<sub>3</sub>O<sub>4</sub> to highly oxidized states

The activity of **KuQ3P**<sub>*n*</sub>@Co<sub>3</sub>O<sub>4</sub> towards photoinduced WO was assayed under 1 Sun illumination using a colloidal suspension of the materials (0.56 mg mL<sup>−1</sup>) in an aqueous NaHCO<sub>3</sub>/Na<sub>2</sub>SiF<sub>6</sub> (pH 5.6) buffer containing 84 mM Na<sub>2</sub>S<sub>2</sub>O<sub>8</sub> as the sacrificial electron acceptor (see details in the ESI, and Fig. S23–S28†). Oxygen evolution was detected during the first 3 h of irradiation for **KuQ3P**<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub> (0.85 μmol mg<sub>material</sub><sup>−1</sup>, with turnover numbers  $\text{TON}_{\text{O}_2/\text{NP}} = 443$  and  $\text{TON}_{\text{O}_2/\text{KuQ}} = 3.8$ , and turnover frequency  $\text{TOF}_{\text{max}} = 1.0$  mol<sub>O<sub>2</sub></sub> h<sup>−1</sup> mol<sub>Co<sub>3</sub>O<sub>4</sub></sub><sup>−1</sup>). Based on our photophysical and electrochemical studies, the proposed photochemical cycle involves reductive quenching of the singlet <sup>1</sup>\***KuQ3P** by the appended Co<sub>3</sub>O<sub>4</sub>, followed by reconversion of **KuQ3P**<sup>•−</sup> to **KuQ3P** upon ET to the persulfate anion (Scheme 3A). The amount of O<sub>2</sub> evolved by the **KuQ3P**<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub> dyad was also greater than that for **KuQ3P**<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub>, suggesting that an excess of **KuQ3P** bound





**Scheme 3** (A) Proposed mechanism for photoinduced WO by  $\text{KuQ3P}_n\text{@Co}_3\text{O}_4$  in a colloidal suspension, employing  $\text{Na}_2\text{S}_2\text{O}_8$  as an irreversible electron acceptor. The preferred reductive quenching of **KuQ** dyes is further supported by the negligible direct reactivity of both singlet and triplet ( $^3\text{KuQ3P}$ ) excited states of the dye with the  $\text{Na}_2\text{S}_2\text{O}_8$  electron acceptor (Fig. S21 and S22†), thus excluding the photoinduced oxidation of **KuQ3P** (oxidative quenching pathway). The **KuQ3P** oxidative radical decomposition pathway likely involved the methylene site on the side chain, at the  $\alpha$  position to the five-membered central ring. (B) Energy levels involved in the initial steps of the WO photocatalytic cycle under our conditions.<sup>90</sup> Note: potentials are not to scale.

at the  $\text{Co}_3\text{O}_4$  NP surface may be precluding the WO process by blocking the active sites.<sup>43</sup> Incidentally, blank experiments conducted with  $\text{Co}_3\text{O}_4^{\text{heptOH}}$  in the absence of **KuQ3P** evolved a modest but non-negligible amount of  $\text{O}_2$  (up to 0.2  $\mu\text{mol}$ , compared with 0.5  $\mu\text{mol}$  evolved by  $\text{KuQ3P}_{0.1}\text{@Co}_3\text{O}_4$ ), consistent with a low photooxygenic activity of unsensitised  $\text{Co}_3\text{O}_4$ .<sup>51,89</sup>

XAS investigation allowed the study of the evolution of  $\text{KuQ3P}_{0.1}\text{@Co}_3\text{O}_4$  along photoirradiation. Interestingly, the XANES data recorded on the photoirradiated material indicated similar features to those for the initial  $\text{Co}_3\text{O}_4^{\text{heptOH}}$  NPs (Fig. 2A). However, a shift of the edge to higher energies was observed, suggestive of a higher oxidation degree of the cobalt centres in comparison with those of  $\text{Co}_3\text{O}_4$ , together with a partial recovery of the intensity in the pre-edge feature. This transformation was found to occur very rapidly (15 s) after starting the irradiation, indicating that the single-atomic Co species quickly evolved under photooxidative conditions. The resulting XANES spectra after irradiation fall between those of  $\text{Co}_3\text{O}_4$  and  $\text{CoO}(\text{OH})$ ,<sup>91,92</sup> the latter being a widely proposed steady-state intermediate for the WO catalytic cycle (Fig. S32†).<sup>93–95</sup>

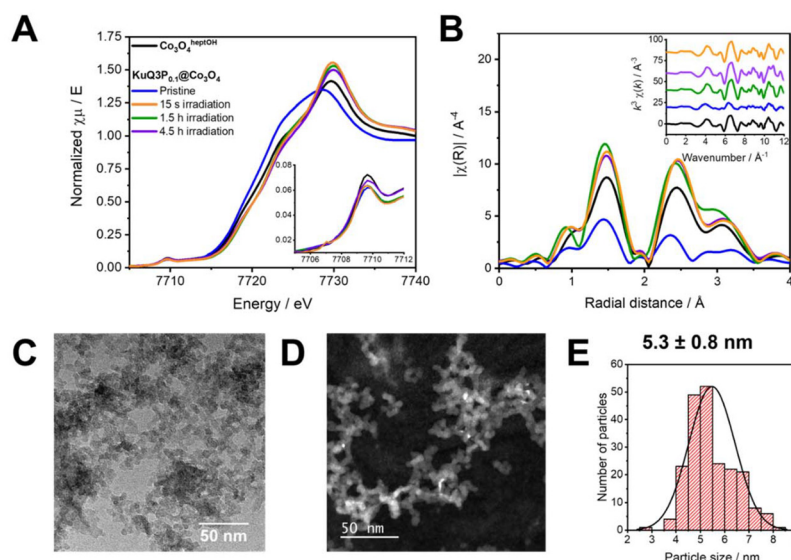
The description of the photoirradiated  $\text{KuQ3P}_{0.1}\text{@Co}_3\text{O}_4$  NPs is further integrated by the recovery of the intensity in the EXAFS spectra, indicating a higher degree of crystallinity of the sample after irradiation (Fig. 2B). Whereas XANES spectra indicate a partial recovery of the  $\text{Co}_3\text{O}_4$  structure, the fittings of the FT-EXAFS spectra revealed that after 15 s of irradiation, the Co–O and Co–Co distances slightly decreased to 1.91 Å and 2.84/3.36 Å, respectively, and further decreased to 1.89 and 2.83/3.34 Å after 4.5 h (Fig. S33, S34 and Tables S5, S6†). These

slightly shortened Co–O and Co–Co bond distances with respect to the 1.92 and 2.86/3.36 Å expected for the pristine  $\text{Co}_3\text{O}_4$  supports the partial transformation of  $\text{Co}_3\text{O}_4$  into octahedrally coordinated  $\text{CoO}(\text{OH})$  species under photochemical conditions.<sup>93</sup>

Visual examination of the reaction mixture at the end of  $\text{KuQ3P}_{0.1}\text{@Co}_3\text{O}_4$  irradiation (after 5 h) showed that the material precipitated during the process to give a black solid. ICP-OES analysis was carried out both on the recovered solid and on the liquid phase. The results indicated a modest leaching of cobalt from  $\text{Co}_3\text{O}_4$  to the solution (6.3%), but a high (67.8%) leaching of phosphorus from **KuQ3P** (Table S4†). Given (i) the colorless nature of the final reaction mixture and (ii) the absence of spectroscopic features of the **KuQ** chromophore upon dichloromethane extraction (Fig. S28†), simple decooordination or loss of the phosphonate moiety by the deeply colored water-soluble **KuQ3P** could be excluded. The experimental evidence instead suggests the chemical disruption of the **KuQ3P** chromophore core, which could originate from the aggressive reaction environment. Indeed,  $\text{SO}_4^{\bullet-}$  radicals generated by  $\text{S}_2\text{O}_8^{2-}$  reductive peroxo cleavage at each photoinduced ET event<sup>96</sup> are powerful oxidizing agents ( $E^\circ$  ranging from 2.6 to 3.1 V vs. the normal hydrogen electrode,

§ Assembly of a dyad equivalent to  $\text{KuQ3P}_{0.1}\text{@Co}_3\text{O}_4$ , could be achieved by co-suspending **KuQ3P** (0.1 eq.) and  $\text{Co}_3\text{O}_4^{\text{heptOH}}$  in the aqueous buffer used for the photoreactions; see Tables S1, S4 and Fig. S26, S27.† In contrast, a co-suspension of  $\text{Co}_3\text{O}_4^{\text{heptOH}}$  and **KuQ-Hex** (incapable of binding to cobalt(II,III) oxide due to the lack of a suitable anchoring group) allows for complete recovery of **KuQ-Hex** upon extraction with dichloromethane, even after photoirradiation in the presence of persulfate (Fig. S28†).





**Fig. 2** (A) Normalized Co K-edge XANES spectra of  $\text{Co}_3\text{O}_4^{\text{heptOH}}$  (black line), pristine (blue line)  $\text{KuQ3P}_{0.1}@\text{Co}_3\text{O}_4$  NPs, and  $\text{KuQ3P}_{0.1}@\text{Co}_3\text{O}_4$  NPs after 15 seconds (orange line), 1.5 h (green line) and 4.5 h (purple line) of photoirradiation. Conditions:  $\text{NaHCO}_3/\text{Na}_2\text{SiF}_6$  (pH 5.6) electrolyte containing 84 mM  $\text{Na}_2\text{S}_2\text{O}_8$ ; irradiation was performed using  $100 \text{ mW cm}^{-2}$  simulated visible solar light ( $\lambda > 400 \text{ nm}$ ). Inset: zoom of the pre-edge region. (B) Fourier transforms of the  $k^3$ -weighted Co EXAFS spectra for the same samples. Inset: corresponding  $k^3$ -weighted Co EXAFS spectra. (C) HR-TEM images of  $\text{KuQ3P}_{0.1}@\text{Co}_3\text{O}_4$  NPs recovered after 5 h of irradiation in the colloidal suspension ( $\text{NaHCO}_3/\text{Na}_2\text{SiF}_6$ , pH 5.6, 0.84 mM  $\text{Na}_2\text{S}_2\text{O}_8$ ). (D) STEM-HAADF images and (E) size distribution histogram of the  $\text{KuQ3P}_{0.1}@\text{Co}_3\text{O}_4$  NPs recovered after 5 h of irradiation.

NHE, for  $\text{SO}_4^{\cdot-}/\text{SO}_4^{2-}$ ), which can react with organic compounds and generate carbon-based radicals. Furthermore,  $\text{SO}_4^{\cdot-}$  can react with water to activate aggressive Fenton chemical processes.<sup>96–99</sup> These considerations support the attribution of the fast deactivation of the colloidal system in photoinduced WO to the intrinsic persulfate reactivity.

Consistent examination by TEM and STEM-HAADF of the solid recovered after 5 h of irradiation revealed the absence of isolated atoms around the  $\text{Co}_3\text{O}_4$  NPs (*vide supra*) accompanied by a diameter increase from  $3.9 \pm 0.6 \text{ nm}$  to  $5.3 \pm 0.8 \text{ nm}$  (Fig. 2C–E and Fig. S29–S31 in the ESI†). This size increase may arise from an Ostwald ripening process prompted by the photoirradiation conditions that induced progressive disruption of  $\text{KuQ3P}$ , together with the incorporation of the single-atomic Co species initially observed in the images of  $\text{KuQ3P}_n@\text{Co}_3\text{O}_4$  hybrid nanomaterials.<sup>100</sup>

Overall, XAS analysis of the hybrid nanomaterials after photoirradiation in the presence of persulfate supports the achievement of a higher overall oxidation state of the  $\text{Co}_3\text{O}_4$ -based NPs, coherent with oxidative activation of the  $\text{Co}_3\text{O}_4$  WOC along with chemical and structural transformations highlighted by HR-TEM and ICP-OES (*i.e.*, disruption of the  $\text{KuQ3P}$  chromophore and NP size increase) during the photoinduced process.

### $\text{KuQ3P}_n@\text{Co}_3\text{O}_4$ on *meso*- $\text{SnO}_2$ for photoelectrochemical water oxidation

Based on the competence of  $\text{KuQ3P}_n@\text{Co}_3\text{O}_4$  dyads in photoinduced WO, we sought to steer their oxygen evolving reactivity in a DSPEC configuration. Thus, we translated the dyadic systems into a regenerative photoanode by embedding them on a  $\text{SnO}_2$  semiconductor film, avoiding the need for the  $\text{S}_2\text{O}_8^{2-}$  electron acceptor and the generation of highly oxidizing radicals. Mesoporous tin oxide (*m*- $\text{SnO}_2$ ) was chosen as the semiconductor due to its low conduction band edge energy, providing more favorable charge injection with respect to other n-type semiconductors.<sup>16,101</sup> Electrodes with a  $2 \mu\text{m}$  *m*- $\text{SnO}_2$  film thickness and 10–40 nm pore size, blade-coated onto FTO conductive slides, were employed in the present work, as they were previously optimized in photoelectrochemical applications of  $\text{KuQ}$  dyes.<sup>67,68</sup>

$\text{SnO}_2$  photoelectrodes functionalized with  $\text{KuQ3P}_n@\text{Co}_3\text{O}_4$ , hereafter indicated as  $\text{SnO}_2|\text{KuQ3P}_n@\text{Co}_3\text{O}_4$ , were obtained by dropcasting a colloidal suspension of the  $\text{KuQ3P}_n@\text{Co}_3\text{O}_4$  NPs in distilled tetrahydrofuran (THF), followed by air drying. Analysis of the XAS and resonance Raman spectra of  $\text{SnO}_2|\text{KuQ3P}_{0.2}@\text{Co}_3\text{O}_4$  showed consistency with those of the as-prepared  $\text{KuQ3P}_{0.2}@\text{Co}_3\text{O}_4$  sample (Fig. S35–S37†).

In order to allow a comparison among the different materials, we deposited an amount of NPs to attain the same nominal  $\text{KuQ3P}$  loading of  $140 \text{ nmol}_{\text{dye}} \text{ cm}^{-2}$ .<sup>67</sup> The photoelectrodes were then tested in a three-electrode photoelectrochemical cell configuration (Fig. S38†), integrating a glassy carbon auxiliary electrode and a  $\text{Ag}/\text{AgCl}$  (3 M  $\text{NaCl}$ ) reference elec-

† When considering the oxidation of  $\text{Co}_3\text{O}_4$  through a photoinduced oxidative cycle, the release of highly oxidizing sulfate radicals can also contribute to the oxidation of  $\text{Co}^{\text{II}}$  sites.



trode (potentials were then converted *vs.* RHE:  $E$  (V *vs.* RHE) =  $E$  (V *vs.* Ag/AgCl) + 0.197 + 0.0592 × pH).

The photoelectrochemical response was analyzed by comparing linear sweep voltammograms (LSV traces) under dark, intermittent (chopped), and continuous illumination, upon irradiation of the electrodes through their back side with the solar spectrum (AM 1.5G, 100 mW cm<sup>-2</sup>) cut with a long-pass filter ( $\lambda > 400$  nm) to avoid direct band gap excitation of SnO<sub>2</sub>. Fig. 3 displays the photoelectrochemical characterization of SnO<sub>2</sub>|**KuQ3P<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub>** as a representative case (additional data are reported in Fig. S39 in the ESI†). Under continuous illumination, an anodic photocurrent response associated with photoelectrochemical WO (*vide infra*) was observed, rising at *ca.* 0.54 V *vs.* RHE, below the thermodynamic value for oxygen evolution,  $E_{\text{O}_2/\text{H}_2\text{O}}^\circ = 1.23$  V *vs.* RHE (underpotential regime; a similar onset potential of 0.55 V *vs.* RHE was previously observed for **KuQ**-sensitized SnO<sub>2</sub> electrodes embedding a tetra-ruthenium polyoxometalate catalyst, Ru<sub>4</sub>POM).<sup>67</sup> The photocurrent density then flattens to a plateau value of *ca.* 10  $\mu\text{A cm}^{-2}$  above 0.84 V *vs.* RHE; the photocurrent profile of chopped-light LSVs resembles the one observed under continuous illumination, with cathodic current spikes being observed upon turning the light off up to 0.84 V *vs.* RHE, indicative of recombination events from the photoinduced charge-separated states.<sup>102</sup> At higher potentials, the spikes disappear due to a higher bias-related driving force for photoinduced electrons collection by the FTO contact.<sup>67,103</sup>

Chronoamperometry (CA) analyses performed with SnO<sub>2</sub>|**KuQ3P<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub>** at 1.14 V *vs.* RHE under continuous illumination produced an initial photocurrent density of *ca.* 35  $\mu\text{A cm}^{-2}$ , evolving to a steady-state photocurrent density ( $J_{\text{ss}}$ ) of 5–10  $\mu\text{A cm}^{-2}$  after a few seconds. The latter was shown to be stable for at least 1 h; after 5.5 h, a 60% photocurrent abatement was observed, mainly due to material detachment from the electrode surface (Table 1 and Fig. 3B and S40†). For the sake of comparison, a  $J_{\text{ss}}$  value of 20  $\mu\text{A cm}^{-2}$  was obtained by combining **KuQ3C** with the Ru<sub>4</sub>POM catalyst, although the latter value was registered at 2 Sun intensity. Concerning the photocurrent behaviour in the early few seconds of irradiation, superimposable CA curves could be achieved in consecutive experiments either by allowing the photoanode to rest until its original open circuit potential (OCP) value (*ca.* 0.74 V *vs.* RHE) or by performing a 60 s CA in the dark at 0.54 V *vs.* RHE (inset of Fig. 3B). Therefore, the initial photocurrent decay may be ascribed to photoanode polarization upon light irradiation rather than physical or chemical degradation of the device.<sup>104</sup>

Detection and quantification of the evolved oxygen was performed through a two-plate generator–collector chronoamperometric method<sup>19,67,105,106</sup> (Fig. 3C, see details in the ESI and Fig. S41–S43†). When held at 1.14 V *vs.* RHE, irradiation of the SnO<sub>2</sub>|**KuQ3P<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub>** electrodes provided a faradaic efficiency for oxygen evolution (FE<sub>O<sub>2</sub></sub>) of 87% (Table 1) (red line in Fig. 3C, and Fig. S43†). The steady-state photocurrent densities ( $J_{\text{ss}}$ ) in CA and FE<sub>O<sub>2</sub></sub> were exploited as key performance indicators to compare the reactivity among the different

materials (Table 1 and Fig. S42†). This included a hybrid nanomaterial with the **KuQ** dye bearing a carboxylate linker (**KuQ3C<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub>**, entry 3 in Table 1; characterization studies are presented in the ESI, Fig. S44–S46†) and “unbound” **KuQ|Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup>** systems (entries 4 and 5 in Table 1), prepared by dropcasting a suitable amount of **Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup>** NPs onto pre-sensitized SnO<sub>2</sub>|**KuQ** electrodes, using both **KuQ3C** and **KuQ3P** (preparation and study of the “unbound” photoanodes are described in the ESI, Section 11 and Fig. S47–S51†).

The data in Table 1 can be discussed according to the following points:

(i) The photocurrent density performance indicator suggests a higher activity for **KuQ3P<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub>** with respect to **KuQ3P<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub>** (entries 1 and 2 in Table 1), while both materials are characterized by a high and consistent FE<sub>O<sub>2</sub></sub> of 88 and 87%, respectively (see Section 12 in the ESI† for additional discussions).

(ii) In the **KuQ<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub>** hybrid nanomaterials, the phosphonate anchoring group provides better results than the carboxylate one in terms of photocurrent density (entries 1 and 3 in Table 1: 11 *vs.* 5  $\mu\text{A cm}^{-2}$  for **KuQ3P<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub>** and **KuQ3C<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub>**, respectively), which can be ascribed in part to the lower stability of the bond between Co<sub>3</sub>O<sub>4</sub> and the carboxylate pendant group with respect to the phosphonate one.<sup>107,108</sup>

(iii) A key comparison comes with the “unbound” systems (entries 4 and 5 in Table 1), for which the most striking difference is a drop of the FE<sub>O<sub>2</sub></sub> value to <50%, indicating a much lower efficiency in translating the photo-accumulated holes to the WO process. These results indicate that the direct chemical interaction between the Co<sub>3</sub>O<sub>4</sub> NPs and the dye shell surrounding their surface drives more efficient management of the ET chain in the dyadic materials.<sup>109,110</sup> \*\* This latter point is particularly relevant in the case of the postulated reductive quenching mechanism (*vide supra*), where the Co<sub>3</sub>O<sub>4</sub> WOC is oxidized by the excited state of the dye, <sup>1\*</sup>**KuQ3P**.<sup>111,112</sup>

The response of SnO<sub>2</sub>|**KuQ<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub>** photoanodes was further analyzed in terms of their photo-action spectrum to unequivocally ascribe the photocurrent to the presence of the **KuQ3P** chromophore. The photoelectrodes were thus irradiated with monochromatic light (wavelength intervals of 10 nm) while recording CA curves at 1.14 V *vs.* RHE. The photo-action spectrum of the SnO<sub>2</sub>|**KuQ3P<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub>** electrodes was constructed in terms of their incident photon-to-current conversion efficiency (IPCE)<sup>113</sup> (see the ESI† for a detailed procedure; experimental setup and raw CA curves are reported in Fig. S53 and S54†). An IPCE value of up to

|| A generally poorly reproducible FE<sub>O<sub>2</sub></sub> was observed in the case of “unbound” photoanodes, with average values of 29%; furthermore, repeated CA curves recorded for SnO<sub>2</sub>|**KuQ3P|Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup>** photoelectrodes provided nearly constant FE<sub>O<sub>2</sub></sub> values, thus not encouraging the hypothesis of *in situ* assembly of the dyads during photoelectrocatalysis (Fig. S51 and S52†).

\*\* Additionally, the **KuQ3P** shell on the surface of **KuQ3P<sub>n</sub>@Co<sub>3</sub>O<sub>4</sub>** is expected to provide a platform for better control over proton exchange in photo-driven WO than the hydrophobic heptOH shell in **Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup>** NPs.





**Fig. 3** (A) LSVs recorded for  $\text{SnO}_2|\text{KuQ3P}_{0.2}@\text{Co}_3\text{O}_4$  under darkness (black, dashed line) and under chopped (red, full line) or continuous (blue, dotted line) illumination, in the  $\text{NaHCO}_3/\text{Na}_2\text{SiF}_6$  (pH 5.8) electrolyte, scan rate  $0.020 \text{ V s}^{-1}$ . (B) Sustained (1 h) CA curves of  $\text{SnO}_2|\text{KuQ3P}_{0.2}@\text{Co}_3\text{O}_4$  recorded at 1.14 V vs. RHE in the  $\text{NaHCO}_3/\text{Na}_2\text{SiF}_6$  (pH 5.8) electrolyte. Inset: 20 s CA curves recorded for the same samples. Multiple CA curves were collected, preceded by 30 s electrode de-polarization at 0.54 V vs. RHE. (C) CA curves recorded on  $\text{SnO}_2|\text{KuQ3P}_{0.2}@\text{Co}_3\text{O}_4$  electrodes in a two-plate generator-collector setup to detect evolved  $\text{O}_2$ . The generator was poised at 1.14 V vs. RHE, the collector was poised at  $-0.36 \text{ V}$  vs. RHE; experiments were performed in the  $\text{NaHCO}_3/\text{Na}_2\text{SiF}_6$  (pH 5.8) electrolyte. (D) IPCE photo-action spectrum (red squares) for  $\text{SnO}_2|\text{KuQ3P}_{0.2}@\text{Co}_3\text{O}_4$  photoanodes, constructed using monochromatic light, superimposed to the absorption spectra of  $\text{SnO}_2|\text{KuQ3C}$  electrodes with the dye in the enolate (orange line) and enol (magenta line) forms. CA curves were recorded at 1.14 V vs. RHE in the  $\text{NaHCO}_3/\text{Na}_2\text{SiF}_6$  (pH 5.8) electrolyte. In (A)–(C), irradiation was performed using  $100 \text{ mW cm}^{-2}$  simulated visible solar light,  $\lambda > 400 \text{ nm}$ .

**Table 1** Photoelectrochemical performance of  $\text{SnO}_2|\text{KuQ}_n@\text{Co}_3\text{O}_4$  and  $\text{SnO}_2|\text{KuQ}|\text{Co}_3\text{O}_4^{\text{heptOH}}$  photoanodes (where  $\text{KuQ} = \text{KuQ3P}$  and  $\text{KuQ3C}$  and  $n = 0.1$  and  $0.2$ )

| # | Electrode                                                         | $\text{KuQ}/\text{Co}_3\text{O}_4$<br>loading<br>( $\text{nmol cm}^{-2}$ ) | $J_{\text{ss}}@1.14 \text{ V vs. RHE}$<br>( $\mu\text{A cm}^{-2}$ ) | $\text{FE}_{\text{O}_2}$ (%) |
|---|-------------------------------------------------------------------|----------------------------------------------------------------------------|---------------------------------------------------------------------|------------------------------|
| 1 | $\text{SnO}_2 \text{KuQ3P}_{0.1}@\text{Co}_3\text{O}_4$           | 140/1400                                                                   | 11                                                                  | 88 <sup>a</sup>              |
| 2 | $\text{SnO}_2 \text{KuQ3P}_{0.2}@\text{Co}_3\text{O}_4$           | 140/700                                                                    | 8                                                                   | 87–95 <sup>b</sup>           |
| 3 | $\text{SnO}_2 \text{KuQ3C}_{0.1}@\text{Co}_3\text{O}_4$           | 140/1400                                                                   | 5                                                                   | 87 <sup>c</sup>              |
| 4 | $\text{SnO}_2 \text{KuQ3C} \text{Co}_3\text{O}_4^{\text{heptOH}}$ | 140/1400                                                                   | 14                                                                  | 45 <sup>d</sup>              |
| 5 | $\text{SnO}_2 \text{KuQ3P} \text{Co}_3\text{O}_4^{\text{heptOH}}$ | 140/1400                                                                   | 8                                                                   | 45 <sup>d,e</sup>            |

<sup>a</sup> The value for  $\text{FE}_{\text{O}_2}$  is reported for experiments run with a 150 s illumination phase. <sup>b</sup> See Fig. S39,† bottom right. The higher value of 95% was observed for longer experiments with  $\text{SnO}_2|\text{KuQ3P}_{0.2}@\text{Co}_3\text{O}_4$ , Fig. S42.† The lower  $\text{FE}_{\text{O}_2}$  registered at 150 s with respect to the almost quantitative value at longer times may be explained by the initial need for multi-electron WOC activation to reach their active state before evolving oxygen. <sup>c</sup> See Fig. S45,† right. <sup>d</sup> See Fig. S49.† <sup>e</sup> See Fig. S50† for repeated experiments and footnote †.

0.44% (at 510 nm, the absorption maximum of **KuQ3P**, see Fig. 3D and Fig. S18, S47†) was determined for  $\text{SnO}_2|\text{KuQ3P}_{0.2}@\text{Co}_3\text{O}_4$ ; this value is more than four times higher

than the one reported for  $\text{SnO}_2|\text{KuQ3C}|\text{Ru}_4\text{POM}$  photoanodes, reaching an IPCE value of up to 0.09% at 490 nm.<sup>67</sup> Additionally, the operational chemical state of the **KuQ** dye in the dyad could be determined from the photo-action spectrum: **KuQ** dyes can in fact partake in enol-enolate equilibria ( $\text{pK}_a \approx 5$ ),<sup>72</sup> with the enol and enolate forms displaying different optical spectra.<sup>67,72</sup> Indeed, the IPCE spectrum suggests that the **KuQ3P** chromophore within the dyad is in its enolate form, supported by a comparison with the absorption spectra of  $\text{SnO}_2|\text{KuQ3C}$  electrodes with the dye in the enolate (Fig. 3D, orange line) and enol (magenta line) forms.††

State-of-the-art systems based on organic dyes for photoelectrochemical WO have been recently surveyed (Table S6†).<sup>3,4,114</sup> These systems are often characterized by photocurrent densities from tenths up to a few hundreds of  $\mu\text{A cm}^{-2}$ , associated with  $\text{FE}_{\text{O}_2}$  in the range of 30–95%, while

†† **KuQ3C** was chosen to compare the optical features of **KuQ** dyes on  $\text{SnO}_2$  films, since its enolate and enol forms are both accessible by chemical means once chemisorbed. **KuQ3P**, on the other hand, directly provides sensitized  $\text{SnO}_2$  featuring the enol form of the dye, as described in the ESI (Sections 2 and 11).†



systems operating at wavelengths above 500 nm are rare. More specifically, in the case of dyadic hybrid nanomaterials, the most pertinent comparisons deal with: (i) the Ru(bpy)<sub>3</sub>-IrOx hybrids proposed by Mallouk and coworkers<sup>33–36,38</sup> (systems A and B in Scheme 1) bound to nanostructured TiO<sub>2</sub>, displaying photocurrent densities between 15 and 80–100  $\mu\text{A cm}^{-2}$ , FE<sub>O<sub>2</sub></sub> between 20 and 100% and IPCE up to 0.7% at 464 nm at pH 5.8, and (ii) the Ru(phenanthroline)-Fe/FeOx material<sup>44</sup> (system D in Scheme 1) onto flat FTO slides, for which photocurrent densities of 10–30  $\mu\text{A cm}^{-2}$  were registered in NaOH solutions at pH 13, although no FE<sub>O<sub>2</sub></sub> or IPCE values were reported. In this broader context, the high FE<sub>O<sub>2</sub></sub> and notable IPCE beyond 500 nm for the Ru-free SnO<sub>2</sub>|KuQ<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub> photoanodes are significant results. An improvement in the performance, and in particular of the photocurrent density, should mainly target appropriate design of the KuQ3P@Co<sub>3</sub>O<sub>4</sub> interface with the surface of the SnO<sub>2</sub> semiconductor. Efforts in this direction should follow literature indications, such as: (i) a further nanostructuring of the semiconductor<sup>115</sup> to increase the hybrid loading, (ii) the use of redox mediators<sup>116,117</sup> – also in polymeric forms<sup>118</sup> – to favour photoinduced electron injection into the semiconductor (a 50-fold increase of the maximum photocurrent was observed for photosystem

II-functionalized photoanodes in the presence of osmium-based polymeric redox mediators), and (iii) the use of an insulating layer to limit back electron transfer.<sup>119</sup> The use of surface coatings (*i.e.*, polymeric or hydrogel coatings)<sup>120</sup> was reported to be beneficial for the stability of molecularly-designed photoanodes.

The evolution of KuQ3P<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub> during photoelectrochemical WO was investigated by complementary resonance Raman and XAS spectroscopy analyses. The resonance Raman spectrum (Fig. 4A) shows that the KuQ3P bands are reasonably intact after 1 h of photoelectrolysis, consistent with the stable photocurrent and preserved oxygen evolution ability of the regenerative photoelectrodes. Therefore, under these operation conditions, the KuQ dye shows good stability; KuQ dyes were indeed shown to be recyclable up to 20 times in aerobic homogeneous photo-oxidation of thioethers.<sup>70</sup>

The major changes in Raman spectra are associated with the A<sub>1g</sub> band of Co<sub>3</sub>O<sub>4</sub>, which undergoes a blue shift from 680 cm<sup>-1</sup> in the pristine KuQ3P<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub> material to 688 cm<sup>-1</sup> in the sample subjected to 1 h of photoelectrolysis. No further changes ascribable to a disorder increase in the material could be identified, thus supporting the absence of particle disruption as a consequence of photoelectrocatalysis. The observed blue shift of the A<sub>1g</sub> mode, on the other hand,



**Fig. 4** (A) Resonance Raman spectra of SnO<sub>2</sub>|KuQ3P<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub> electrodes before (red line) and after (purple line) 1 h of photoelectrolysis at 1.14 V vs. RHE in the NaHCO<sub>3</sub>/Na<sub>2</sub>SiF<sub>6</sub> (pH 5.8) electrolyte. The full spectral region is displayed. Inset: zoom of the resonance Raman spectra in the 400–750 cm<sup>-1</sup> region, displaying changes in signals pertaining to Co<sub>3</sub>O<sub>4</sub> and CoO(OH). (B) Normalized Co K-edge XANES spectra of Co<sub>3</sub>O<sub>4</sub><sup>heptOH</sup> NPs (black line) and SnO<sub>2</sub>|KuQ3P<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub>, before (red line) and after 15 s (green line) and 30 min (purple line) of photoelectrocatalysis at 1.14 V vs. RHE in the NaHCO<sub>3</sub>/Na<sub>2</sub>SiF<sub>6</sub> (pH 5.8) electrolyte. Inset: zoom of the edge (top) and pre-edge (bottom) regions. (C) Fourier transforms of the *k*<sup>3</sup>-weighted Co EXAFS spectra for the same samples (*k*-range of 3–11 Å<sup>-1</sup>). Inset: corresponding *k*<sup>3</sup>-weighted Co EXAFS spectra; same color code as in (B).



could be attributed to an increasing component of the  $A_{1g}$  mode of  $\text{CoO}(\text{OH})$  formed during photoelectrolysis. This hypothesis is supported by the presence of the bands at  $471\text{ cm}^{-1}$  and  $508\text{ cm}^{-1}$ , attributed to the  $\text{Co}_3\text{O}_4$   $F_{2g}$  and  $E_g$  modes, respectively, that evolve into a weaker signal centered at  $485\text{ cm}^{-1}$ . Albeit ill-defined, this novel feature is coherent with the  $E_g$  mode of  $\text{CoO}(\text{OH})$ .<sup>48</sup> Therefore, resonance Raman surface probing of the dyads after extensive photoelectrochemical oxygen evolution has led to the identification of a modification of the Co-based WOC, evolving from a preeminently  $\text{Co}_3\text{O}_4$  structure towards a mixture of  $\text{Co}_3\text{O}_4$  and  $\text{CoO}(\text{OH})$  coexisting on the surface of the NPs, consistent with XAS data analysis (*vide infra*).

XAS spectra were acquired *ex situ* on  $\text{SnO}_2|\text{KuQ3P}_{0.2}@\text{Co}_3\text{O}_4$  subjected to photoelectrocatalysis for up to 30 min (Fig. S55 and S56†), under the same conditions reported above (Fig. 4B and C). Monitoring of XAS features of  $\text{SnO}_2|\text{KuQ3P}_{0.2}@\text{Co}_3\text{O}_4$  photoanodes subjected to increasing photoelectrolysis time revealed significant changes that can be summarized as follows:

(i) In the XANES spectra, the edge is gradually shifted towards higher energy values, consistent with an enrichment of  $\text{Co}^{\text{III}}$  over  $\text{Co}^{\text{II}}$  sites.

(ii) The intensity of the pre-edge region of the spectra remains unchanged after the catalytic process, indicating that the species maintained a centrosymmetric  $O_h$  geometry around the metal center.

(iii) In the FT-EXAFS spectra, a recovery of the  $\text{Co}_3\text{O}_4$  structure can be observed, accompanied by a very minor shortening of the Co–Co distances (Fig. S4A and S56;† fitting parameters are collected in Table S7†).

Overall, XAS analysis shows a similar trend in comparison with the colloidal system, indicating a partial evolution of the spinel  $\text{Co}_3\text{O}_4$  material into a  $\text{Co}_3\text{O}_4/\text{CoO}(\text{OH})$  species under catalytic conditions, consistent with our Raman data and existing reports.<sup>95,121</sup> Additional support for the formation of  $\text{CoO}(\text{OH})$  as a result of WO is provided by CV experiments (Fig. S57†).<sup>122</sup> The role of the  $\text{CoO}(\text{OH})$  species in WO has been indeed documented in the literature.<sup>56,123</sup>

## Conclusions

This work describes the preparation of the first fully noble metal-free dyadic hybrid nanomaterial constituted by  $\text{Co}_3\text{O}_4$  nanoparticles functionalized with a fully organic molecular dye, able to catalyze visible light-induced WO. Our strategy takes advantage of the organometallic synthesis to achieve nanometer-sized, monodisperse Co NPs, which were controllably oxidized and covalently post-functionalized with a **KuQ** dye *via* a phosphonate anchoring group. **KuQ3P** was confirmed to act as a LHU within the coordination sphere of the  $\text{Co}_3\text{O}_4$  NPs, driving WOC activation *via* a mechanism suggestive of directional photoinduced ET.  $\text{SnO}_2$  photoanodes then provided a suitable platform for the dyads, displaying selective  $\text{O}_2$  evolution in a visible light-operated DSPEC. Spectroscopic evi-

dence allowed us to assess the stability of both the  $\text{Co}_3\text{O}_4$  WOC unit and the **KuQ3P** LHU, and to gain insight into the WO mechanism.

Current studies are being devoted to the implementation of **KuQ3P<sub>n</sub>@Co<sub>3</sub>O<sub>4</sub>** dyads into higher complexity DSPEC devices tackling greater oxygenic photocurrents by means of suitably designed redox mediators or semiconductor–molecule heterojunctions. The approach herein described introduces to a wider scenario, where the principles relevant to artificial photosynthesis, such as photoinduced ET preferential pathways, could be routinely interpreted by modular design strategies applied to functional nanomaterials starting at the molecular level.

## Experimental

### General methods

For the synthesis of the  $\text{Co}^{\text{heptOH}}$  NPs, operations were conducted using Schlenk-line techniques or in a glovebox, employing Ar as the inert gas. (Cyclooctadienyl)(1,5-cyclooctadiene) cobalt(i),  $[\text{Co}(\text{COD})(\text{COE})]$ , was purchased from Nanomeps. Sodium persulfate, sodium hydroxide, sodium hexafluorosilicate, and sodium hydrogen carbonate were acquired from Merck. 1-Heptanol (heptOH) was dried on 4 Å molecular sieves, filtered using a cannula, and degassed by Ar bubbling before the synthesis. Pentane was dried using an MBraun solvent purifier, passed through molecular sieve columns, and degassed by three freeze–pump–thaw cycles before use. Ultrapure argon was purchased from Abelló Linde. Tetrahydrofuran (THF) was purified by distillation over sodium-benzophenone under an Ar atmosphere, subsequently degassed by three freeze–pump–thaw cycles and stored in an Ar-filled glovebox. 1-Heptanol-stabilized  $\text{Co}_3\text{O}_4$  NPs ( $\text{Co}_3\text{O}_4^{\text{heptOH}}$ ) and **KuQ** dyes were prepared following previously reported protocols.<sup>43,66–68,70</sup>

### Synthesis of **KuQ3P<sub>n</sub>@Co<sub>3</sub>O<sub>4</sub>** NPs

A colloidal suspension of  $\text{Co}_3\text{O}_4^{\text{heptOH}}$  NPs (10 mg) was prepared in heptOH (0.1 mL) by sonication. Independently, a suspension of 1-[3-(dihydroxyphosphonyl)propyl]**KuQuinone** (**KuQ3P**) (0.78 mg or 1.55 mg, corresponding to 0.10 or 0.20 equivalents, respectively) was prepared in methanol (2.4 mL) by sonication. Both suspensions were mixed, and water (0.4 mL) was added. Each mixture was stirred for 4 days in the dark, then centrifuged and washed with methanol and hexane, and dried under vacuum, yielding a brown powder corresponding to **KuQ3P<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub>** NPs or **KuQ3P<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub>** NPs, respectively. All materials were characterized by TEM in order to determine the NP mean size. ICP-OES or ICP-MS allowed the determination of the composition. **KuQ3P<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub>** NPs:  $3.9 \pm 0.6\text{ nm}$ , 35% Co, 0.52% P. **KuQ3P<sub>0.2</sub>@Co<sub>3</sub>O<sub>4</sub>** NPs:  $3.0 \pm 0.6\text{ nm}$ , 35% Co, 1.02% P.

The same protocol was applied using 0.10 equivalents of 1-(3-carboxylpropyl)**KuQuinone** (**KuQ3C**) to obtain **KuQ3C<sub>0.1</sub>@Co<sub>3</sub>O<sub>4</sub>** NPs. The only difference registered was an orange intermediate suspension, observed after 1 h, which



then turned into a brown suspension.  $\text{KuQ3C}_{0.1}\text{@Co}_3\text{O}_4$  NPs:  $3.7 \pm 0.6$  nm, 35% Co.

Full details of microscopic and spectroscopic characterization of the systems, procedures for the preparation and characterization of the photoanodes, and photocatalytic and photoelectrochemical experiments are presented in the ESI.†

## Author contributions

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

## Data availability

Data will be made available upon reasonable request to the corresponding authors. The raw synchrotron data are available at the Edmond repository.

## Conflicts of interest

There are no conflicts of interest to declare.

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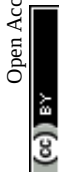
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