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Metal–organic-framework embellished through ion etching method for highly enhanced electrochemical oxygen evolution reaction catalysis†

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Metal–organic-frameworks (MOFs) are scarcely considered to catalyse electrochemical reactions directly due to the limitation of their bulk structure, poor conductivity, and scarce active sites. Fe ions, as effective activity boosters for nickel-based catalysts, can effectively modulate the neatly arranged structure and electron configuration of MOF materials through the penetrating and etching process. Therefore, a Fe ion etching method was employed to modulate the structure and electronic configuration of bulk MOF. The resulting Ni-MOF-Fe-2 was equipped with a wider pore width distribution and lower crystallinity, and the Fe ion doping induced many dislocations and stacking faults in the lattice planes, which provided sufficient defects and active sites for OER. Therefore, Ni-MOF-Fe-2 displayed an advanced performance with an overpotential of 269 mV at the current density of 10 mA cm⁻², and the small Tafel slope of 47.1 mV dec⁻¹ and charge transfer resistances (R_{ct}) of 8.31 Ω revealed its fast kinetics and high electron transfer efficiency, indicating that Fe ion etching played an important role in booting OER performance of Ni MOF.

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Introduction

With the increasing energy crisis and environmental pollution, developing new technology for efficient and environment-friendly energy conversion and storage is crucial.^{1,2} Hydrogen (H₂), the cleanest fuel available, is the optimal candidate to meet these needs of future society due to its high energy density and energy conversion efficiency.^{3–5} Among the various methods of hydrogen production, water splitting has attracted increasing attention due to its abundant feedstock, high product purity, environmental friendliness, and considerable sustainability.^{6–9} Although the water splitting devices are made to produce hydrogen rather than oxygen, the oxygen evolution reaction (OER) has become a dominant factor limiting the efficiency of water oxidation due to its sluggish kinetics.^{10–12} In the early

stages, proton exchange membrane (PEM) water electrolyzers and alkaline water electrolyzers both used noble metal oxides IrO₂ and RuO₂ to catalyze OER, but they are limited by high cost and their instability under acidic conditions.^{13,14} Therefore, developing efficient, earth-abundant and stable OER catalysts has become an urgent global problem in recent years.

Metal organic frameworks (MOFs) are a kind of crystalline materials with periodically arranged network structure.^{15–17} Their porous structure facilitates the penetration of electrolytes and the accessibility of reaction substances, which is ideal for transition metal carriers and efficient catalysts.^{18,19} In general, however, MOF has poor conductivity and poor stability at the voltages required for electrochemical reactions.²⁰ Thus, derivatives of transition metal-based MOFs are considered ideal alternatives due to their promising catalytic activity, outstanding stability, and abundant resources, and these include oxides, sulfides, nitrides, phosphides, *etc.*^{21–25} Moreover, transition metal compounds are considered to achieve easier electron transfer due to multiple oxidation states,^{26–28} thus enhancing catalytic activity. There are few reports about MOFs acting as electrochemical catalysts directly. In order to promote their electrical conductivity, MOFs are combined with conductive carriers to enhance the electrical conductivity of MOF, including nickel foam,^{29,30} carbon dots,^{31,32} and graphene.^{33,34} Among these MOFs, nickel-based

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MOFs have attracted increasing attention due to Ni ions' inherent OER catalytic activity.^{35–37} Using MOFs as electrocatalysts directly will be a significant and highly desirable challenge for this project.

Herein, the Fe ion etching of nickel based bulk MOF, Ni-MOF-Fe-2, was designed and tested. The introduction of iron ions into nickel-based catalysts can substantially contribute to OER performance. Iron ions, as a kind of Lewis acid, can generate H^+ through hydrolysis, which can penetrate and etch the originally neatly arranged structure of MOF during the reaction, and it is an effective means to regulate the morphology of MOF-based catalysts.^{38–40} Therefore, it is a promising approach to design remarkable catalysts that boosts OER performances by doping iron into nickel-based catalysts through the solvothermal process. Compared with pristine MOF, the Fe ion doped sample featured lower crystallinity caused by the deformed lattice plane, resulting in various defects that facilitated its OER activities. XPS results showed the Fe ion modified catalysts were loaded with higher chemical valence state nickel ions, indicating its superior OER activity. As a result, Ni-MOF-Fe-2 delivered a current density of 10 mA cm^{-2} with an overpotential of only 269 mV, and it also showed excellent stability. EIS and C_{dl} results illustrated the Ni-MOF-Fe-2 was more favorable in conductivity and active sites, and the method provided a feasible attempt to promote OER activities of future MOF-based catalysts.

Experimental

Materials and chemicals

The chemicals used during the synthesis of catalysts (including nickel(II) nitrate hexahydrate $[Ni(NO_3)_2 \cdot 6H_2O]$, iron(III) chloride hexahydrate $(FeCl_3 \cdot 6H_2O)$, formic acid $(HCOOH)$, and *N,N*-dimethylformamide (DMF)) were purchased from Sinopharm Chemical Reagent Co., Ltd. Nafion solution (5 wt%) was purchased from Sigma-Aldrich. All the chemicals were directly used without any further refinement.

Synthesis of Ni MOF

The precursor Ni MOF was synthesized with reference to previous work.⁴¹ 3.402 g nickel(II) nitrate hexahydrate was dissolved in 15 mL DMF under ultrasonication. After adding 4.5 mL formic acid, the whole system was transferred into a 50 mL Teflon-lined autoclave and sealed, followed by a 12 h solvothermal treatment at 100 °C. After cooling down to room temperature, the pale green product was washed with ethyl alcohol 3 times and dried at 80 °C to prepare for the next step.

Synthesis of Ni-MOF-Fe-x

The catalysts, Ni-MOF-Fe-x, were obtained by etching Ni MOF with different concentrations of iron ion solution. Consider Ni-MOF-Fe-2, for instance. 0.162 g iron(III) chloride hexahydrate was dissolved in 30 mL DMF to form a 0.02 M Fe^{3+} solution. Then, 50 mg of the precursor Ni MOF was added to the solution, transferred to a 50 mL Teflon-lined autoclave, and heated for 4 h at 100 °C in the oven. The product was washed

with ethyl alcohol 3 times and dried at 80 °C. Ni-MOF-Fe-1 and Ni-MOF-Fe-3 were synthesized in the same way, except for changing the concentration of Fe^{3+} solution to 0.01 M and 0.03 M, respectively.

Electrochemical measurements

All the electrochemical measurements were carried out on a CHI-760E electrochemical workstation with a three-electrode system in 1 M KOH. The catalyst, Pt, and Hg/HgO were chosen as the working electrode, counter electrode, and reference electrode, respectively. The catalyst ink was prepared by dispersing 5 mg catalyst in 1 mL Nafion solution (0.1 wt%) by a 30 min ultrasonication. Before loading the catalyst, the surface of the glassy carbon electrode (GCE, diameter = 5 mm) was polished with 1.5 μm , 500 nm, and 50 nm $\alpha\text{-Al}_2\text{O}_3$ successively. After washing with ultrapure water and ethanol, 12 μL ink was dripped on the clean surface and dried at room temperature. The mass loading of the catalyst was calculated as 0.3 mg cm^{-2} .

To remove the unstable parts and activate the catalysts, 200 cyclic voltammetric scans (CVs) from a potential of 1.02 V to 1.82 V (vs. RHE) with a scan rate of 500 mV s^{-1} were taken prior to the measurement. The OER cyclic voltammetry (CV) scans were measured from the potential of 1.12 V to 1.77 V (vs. RHE) at the scan rate of 5 mV s^{-1} , and the polarization curves were determined by the negative scan part of CV, and the overpotentials (η) were obtained from $\eta = E(\text{vs. RHE}) - 1.23\text{ V}$. All the curves were recorded without *iR*-correction.

The Tafel plots of catalysts were depicted from their corresponding linear sweep voltammograms (LSV) data, which can be fitted based on the equation below:

$$\eta = a + b \times \log(j)$$

where η is the overpotential and j is the steady-state current density to the geometric area of the working electrode. The durability test was performed using chronopotentiometric measurements, and a carbon rod was used as the counter electrode to avoid Pt migration. All potentials were calibrated to the reversible hydrogen electrode (RHE).

The electrochemical surface area (ECSA) was evaluated by the CV technique in a low potential range from 1.17 to 1.27 V (vs. RHE) under different scan rates from 10 to 50 mV s^{-1} to avoid the generation of faradic current. The double-layer (C_{dl}) is linearly proportional to the ECSA, and it was estimated by plotting half of the capacitive currents at 1.222 V (vs. RHE) ($\Delta j/2$, Δj is $J_{\text{anodic}} - J_{\text{cathodic}}$) against scan rate. *Operando* electrochemical impedance spectroscopy (EIS) measurements were performed from 0.01–1 000 000 Hz, with the potential ranging from 1.495 to 1.595 V (vs. RHE).

Characterization

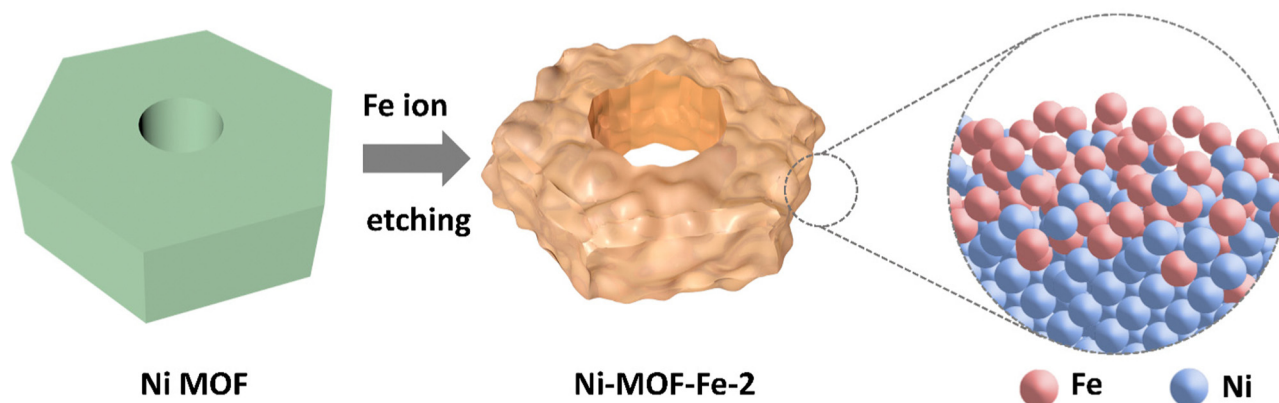
Scanning electron microscopy (SEM) images were obtained on a Zeiss Sigma scanning electron microscope. The high-resolution transmission electron microscopy (HRTEM), the corresponding selected area electron diffraction (SAED), and energy dispersive X-ray spectroscopy (EDX) results were measured on FEI Tecnai G2 F20 field emission transmission microscope. Powder X-ray

diffraction (PXRD) measurements were performed using a Bruker D8 Advance X-ray diffractometer with a Cu K α radiation source ($\lambda = 0.15418$ nm) from 5 to 80° at the scan rate of 5° min⁻¹. N₂ adsorption-desorption analysis was conducted on a Micromeritics ASAP 3020 instrument at 77 K. X-ray photoelectron spectroscopy (XPS) analyses were carried out using an X-ray photoelectron spectrometer (Thermo Fischer, ESCALAB Xi +), and the data were analyzed by CasaXPS Software version 2.3.19.

Results and discussion

Two steps were involved in the fabrication of Ni-MOF-Fe-2, as illustrated in Scheme 1. First, the precursor Ni MOF was synthesized through a solvothermal method. Then, the as-prepared Ni MOF was etched by 0.02 M iron ion(III) dissolved in DMF for 4 h. For comparison, catalysts etched by other concentrations of iron ion(III) (0.01 M and 0.03 M) were prepared in the same way. SEM images in Fig. 1a and TEM images in Fig. S1 (ESI†) showed the precursor Ni MOF consists of bulk MOF with a size of about 500 nm width and 200 nm thickness, equipped with a relatively smooth surface, which was consistent with the morphology in a previous report.⁴² After Fe ion etching, the size of Ni-MOF-Fe-2 was almost unchanged, accompanied by polished edges and rough surface of the bulk structure, which was caused by Fe ions etching the pristine MOF (Fig. 1b). SEM images in Fig. S2 (ESI†) showed Ni-MOF-Fe-2 was corroded by Fe ions and enlarged the cavity in the center of the structure. To explore the etching process, SEM images of Ni-MOF-Fe-1 and Ni-MOF-Fe-3 are displayed in Fig. S3 (ESI†). The morphology of Ni-MOF-Fe-1 fell between that of Ni MOF and Ni-MOF-Fe-2, where the surface began to get rough, and the cavity in the center of the structure did not enlarge as much as Ni-MOF-Fe-2. On the other hand, Ni-MOF-Fe-3 was etched into a jumbled mass of nanoparticles and could not recognize the original MOF block, arising from the excessive concentration of Fe ions. Therefore, it is reasonable to speculate that optimal catalysts require both maintenance of the Ni MOF bulk structure and appropriate etching. Fig. 1b and c both displayed that there were some spherical nanoparticles with the size of about 100 nm randomly

distributed on the surface of the MOF structure, which exhibited amorphous property, as shown in Fig. 1e. Moreover, the high-resolution transmission electron microscopy (HRTEM) in Fig. 1d revealed the matrix of the Ni-MOF-Fe-2 was full of nanocrystalline domains with a size of about 5 nm, which can be distinctly recognized as lattice fringes with the lattice distance of 0.21 nm and 0.23 nm (Fig. 2f, the corresponding contrast intensity profile is displayed in Fig. S4, ESI†), and are identical to the (-121) and (-120) planes of nickel oxide (PDF#14-0481). The selected area electron diffraction (SAED) pattern in Fig. 2f shows the diffraction rings can be indexed to the (-121), (-120), (022), and (023) planes of nickel oxide (PDF#14-0481), as well as the (-122) and (004) planes of nickel hydroxide (PDF#14-0117), which gave the evidence to the nature of the material. Further investigation was adopted for the dispersive nanocrystalline domains. Fig. 2g gave the lattice image from Fig. 2f, and the inset was its corresponding FFT (Fast Fourier Transform) pattern. It can be clearly distinguished that there are two lattice planes corresponding to (-121) and (-120), which can be matched with the SAED pattern in Fig. 2f. IFFT (Inverse Fast Fourier Transform) imaging technique was employed to further characterize the structural defects in this area. Fig. 2h and i show the IFFT images of (-121) and (-120) planes, respectively. They both showed many dislocations and stacking faults (red “T”) around the center area, which was caused by the Fe doping. The different atomic radius and electron configuration between the host element and guest element will inevitably cause the formation of dislocations and stacking faults, developing into defects that boost the OER process.^{43,44} Therefore, the unique widespread but closely adjacent nanocrystalline domains made the crystalline and amorphous regions integrate into Ni-MOF-Fe-2, ensuring efficient charge transfer while providing the catalyst with rich active sites.⁴⁵ High-angle annular dark field scanning transmission electron microscopy (HAADF-STEM) images in Fig. S5 (ESI†) and the corresponding elemental mappings in Fig. 2j displayed that the Ni element was mainly distributed in the MOF matrix, while Fe element mainly existed on the surface and slightly diffused into the MOF matrix, which was consistent with the above speculation of the Fe ion etching process. It is not difficult to



Scheme 1 The fabrication of Ni-MOF-Fe-2.

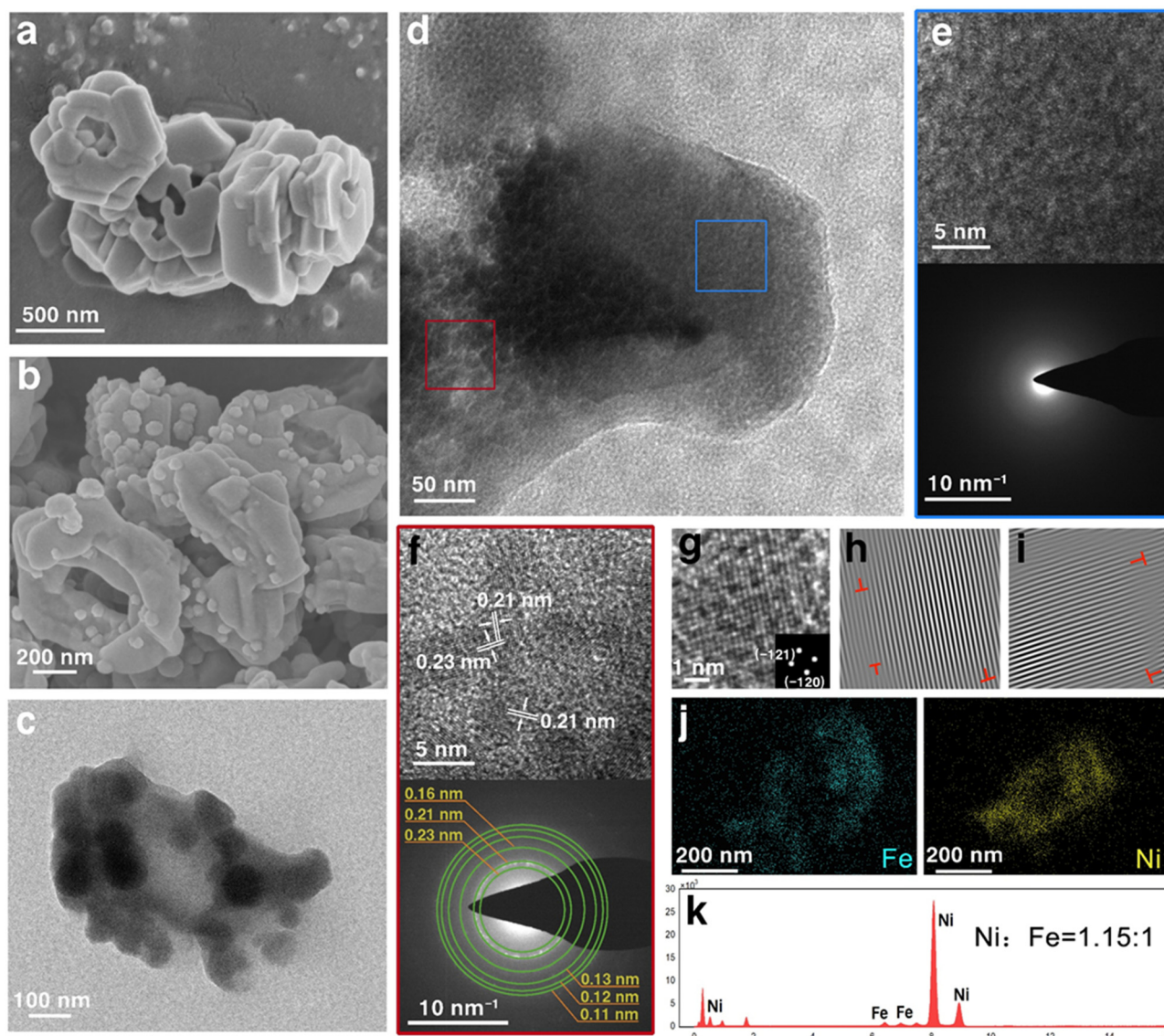


Fig. 1 (a) SEM images of Ni MOF. (b) SEM, (c and d) TEM images of Ni-MOF-Fe-2. (e and f) High resolution TEM images and the corresponding SAED in (d). (g) Atomic lattice image in (f), inset: FFT pattern. (h) Atomic lattice image of (g) imaged along the (-121) lattice plane, and (i) atomic lattice image of (g) imaged along the (-120) lattice plane. (j) Elemental mapping of Fe and Ni in Ni-MOF-Fe-2. (k) EDX results of Ni-MOF-Fe-2.

speculate that the spherical nanoparticles dispersed on the surface of the matrix consist of amorphous iron-based compounds, and a further X-ray photoelectron spectroscopy (XPS) result of Fe 2p in Fig. S6 (ESI[†]) showed the chemical states, with the peaks of Fe 2p_{3/2} mainly located at 706.8 and 710.9 eV, demonstrating the coexistence of Fe²⁺ and Fe³⁺, which revealed the ferric oxide nature of those spherical nanoparticles.^{46–48} Energy dispersive X-ray spectroscopy (EDX) results in Fig. 1k and Table S1 (ESI[†]) gave the atomic ratio of Ni/Fe as 1.15:1, which further confirmed the co-existence of Ni and Fe, illustrating the successful introduction of Fe.

To further investigate the process of the MOF material structure evolution, powder X-ray diffraction (PXRD) was applied to study the crystal structure. As shown in Fig. 2a, lines I, II, III, and IV belonged to Ni MOF, Ni-MOF-Fe-1, Ni-MOF-Fe-2,

and Ni-MOF-Fe-3, respectively, and all the peaks of each sample were well fitted with $[\text{Ni}_3(\text{HCOO})_6]\cdot\text{DMF}$ reported by Li *et al.*,⁴¹ which verified the successful synthesis of target MOF.

The difference was that the peak intensity of the samples after Fe ion etching got weaker and decreased with the increase of iron ion concentration. It suggested that the Fe ions could etch samples effectively and cause crystal amorphization, which formed defects in the crystal and, in turn, became the active sites during the reaction process.⁴⁹

N₂ absorption and desorption isotherms were employed to ascertain the specific surface area and pore size distribution of Ni MOF and Ni-MOF-Fe-2. In Fig. 2b and c, the two samples showed similar isotherms, suggesting that the specific surface area and pore structure of the products were well maintained under the action of suitable concentration of iron ions, and the

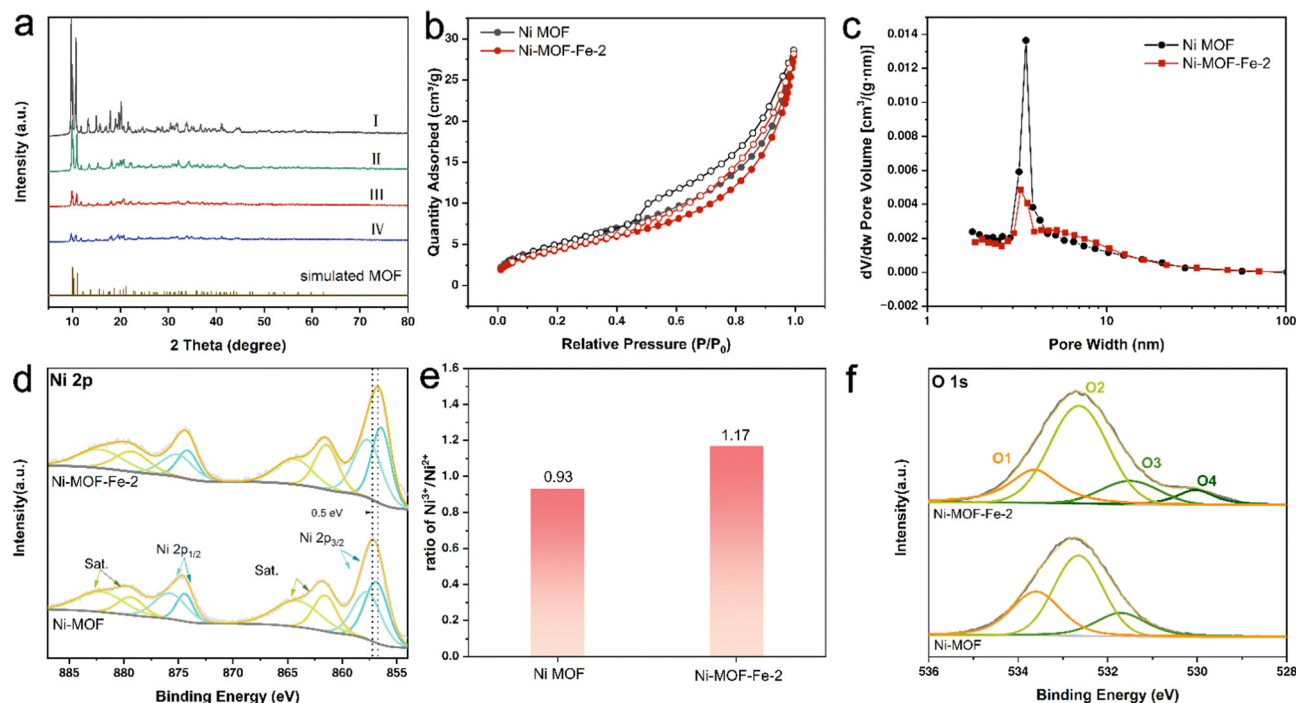


Fig. 2 (a) XRD patterns tested and simulated, (b) N₂ absorption-desorption isotherms, (c) pore width distribution, (d) high-resolution XPS spectra of Ni 2p, (e) ratio of Ni³⁺/Ni²⁺ and (f) high-resolution XPS spectra of O 1s of Ni MOF and Ni-MOF-Fe-2.

measured specific surface areas of Ni MOF and Ni-MOF-Fe-2 were 19.1 and 16.2 m² g⁻¹, and their pore volumes were 0.042 and 0.041 cm³ g⁻¹, respectively. The pore width distribution in Fig. 2c showed the pores in Ni MOF concentrated on 3.5 nm, which featured the common mesoporous MOF,⁵⁰ while the pore width of Ni-MOF-Fe-2 got more emanative and the proportion of the pore width between 4–10 nm increased compared to Ni MOF (Fig. S7, ESI†). Therefore, the specific surface area and pore volume of Ni-MOF-Fe-2 were basically maintained after etching, and some larger pores were introduced, facilitating the exposure of active sites and mass transfer at the catalytic interface.

X-ray photoelectron spectroscopy was adopted to study the chemical states and electronic interactions of the main elements before and after Fe ion etching. Fig. S8 (ESI†) gave high-resolution XPS of C 1s of Ni MOF and Ni-MOF-Fe-2, which can be fitted to the peaks located at 284.8, 285.8, and 289.2 eV and assigned to C–C, C–N, and –COOH species, respectively.^{9,42} Ni 2p spectra in Fig. 2d showed the peaks centered at 856.7 and 857.6 eV arising from Ni²⁺ and Ni³⁺.^{39,51,52} After etching, the ratio of Ni³⁺/Ni²⁺ increased from 0.93 to 1.17 (Fig. 2e), indicating that strong electronic interactions occurred between Ni and Fe and higher chemical valence state nickel ions were produced, which was more favourable to boosting the OER catalytic activity.^{52,53} Moreover, the O 1s spectra were deconvoluted into peaks centered at 533.7(O1), 532.6(O2), and 531.3(O3) eV in Fig. 2f, which were caused by water molecules adsorbed on the surface, hydroxyl groups (–OH), and lattice oxygen.^{54,55} It is observed that a peak centered at 529.7 eV emerged in Ni-MOF-Fe-2, which can be assigned to the oxygen vacancy (Ov) resulting

from the Fe doping treatment.^{56,57} Compared with Ni MOF, the spectrum of Ni 2p in Ni-MOF-Fe-2 showed a 0.5 eV negative shift, indicating that the electronic configuration of Ni has been modulated by the doped Fe, and strong electronic interaction occurred between the metal ions.⁵⁸ Thus, more active sites were introduced and the OER performance was significantly enhanced.

The OER activity of the as-prepared catalysts was evaluated in 1.0 M KOH solution and benchmarked against that of bare Ni MOF and commercial IrO₂ catalyst, and all the curves were collected without *iR*-correction. Encouraged by the introduction of Fe, Ni-MOF-Fe-2 exhibited an advanced OER performance. Fig. 3a and b show the evolution CV curves of Ni MOF and Ni-MOF-Fe-2, revealing the activation process of these two samples. Note that the intensity of the Ni²⁺/Ni³⁺ redox peak increased with the CV cycles and tends to be stable. Compared with the unchanged cathodic peak in Ni MOF catalyst, the cathodic peak of Ni-MOF-Fe-2 showed an evident negative shift (–39 mV) from the first scan to the 100th one, accompanied by the increased cut-off current density (Fig. S9, ESI†). The phenomenon originated from the introduction of iron ions, which regulated the local electronic environment of Ni, facilitating the catalyst more advantageous for the OER process,^{59–61} rather than simply through the transient current increase caused by the oxidation of nickel ions, which was unstable and descending. Therefore, the as-prepared Ni-MOF-Fe-2 possessed outstanding OER performances. As shown in Fig. 3c, the LSV curve of Ni-MOF-Fe-2 overwhelmed that of the precursor Ni MOF and commercial IrO₂, demonstrating that the unique iron ion etching was highly favourable for the OER process, and this was supported by the

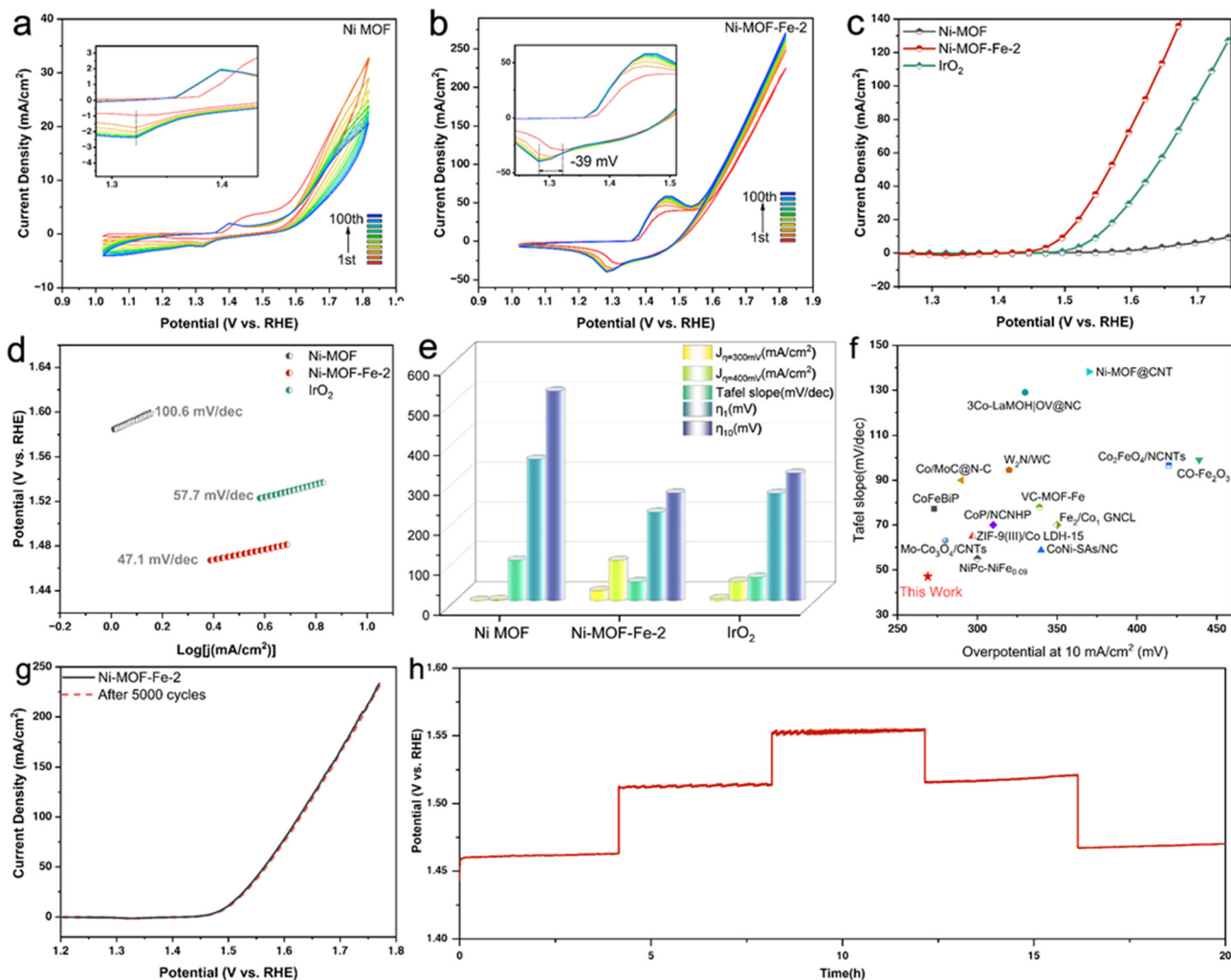


Fig. 3 OER performances. The evolution of CV curves for (a) Ni-MOF and (b) Ni-MOF-Fe-2 from the 1st to the 300th scan at 500 mV s^{-1} , inset: enlarged curves of redox peaks at around 1.3 V (vs. RHE). (c) LSV OER polarization curves of the catalysts without iR -correction. (d) Corresponding Tafel slopes in (c). (e) Comparison of overpotentials at 1 mA cm^{-2} (η_1) and 10 mA cm^{-2} (η_{10}), the current densities under overpotentials of 300 mV and 400 mV , and the Tafel slopes obtained from the catalysts. (f) The OER activity comparison from the literature showing the overpotential at 10 mA cm^{-2} with Tafel slope. (g) Polarization curves of Ni-MOF-Fe-2 at the 1st and 5000th cycle for OER. (h) Chronopotentiometric curve of Ni-MOF-Fe-2 at 5 mA cm^{-2} , 20 mA cm^{-2} , and 40 mA cm^{-2} .

Tafel slopes in Fig. 3d and comparison in Fig. 3e, where the Ni-MOF-Fe-2 can deliver a current density of 10 mA cm^{-2} with an overpotential of only 269 mV , outperforming that of commercial IrO_2 (319 mV) and significantly exceeding bare Ni-MOF (524 mV). Meanwhile, Ni-MOF-Fe-2 showed the largest current density of 25.15 mA cm^{-2} and 100.1 mA cm^{-2} at the overpotential of 300 mV and 400 mV , respectively, which were almost 4.9 times and 2.1 times that of IrO_2 , and even 79.6 times and 37.8 times that of Ni-MOF under the same condition. The smallest Tafel slope of the target sample (47.1 mV dec^{-1} for Ni-MOF-Fe-2, 57.7 mV dec^{-1} for IrO_2 , $100.6 \text{ mV dec}^{-1}$ for Ni-MOF) verified its intrinsically enhanced catalytic activity and kinetics, illustrating the significant role of Fe ion etching in boosting OER performance. More than that, such Tafel slope and overpotential of Ni-MOF-Fe-2 were lower than most of the previously reported powder catalysts, as listed in Fig. 3f and Table S2 (ESI[†]). Furthermore, we found that the catalytic activities of the materials depended on

the concentration of Fe ions. As shown in Fig. S10 (ESI[†]), the catalytic activity of the products exhibited a volcano-type variation as the concentration of iron ion etching increased. The precursor Ni-MOF showed an obvious improvement in catalytic activities under the 0.01 M Fe ion etching effect (Ni-MOF-Fe-1, $\eta_{10} = 294 \text{ mV}$, Tafel slope = 52.7 mV dec^{-1}), and this positive promotion reached the maximum for Ni-MOF-Fe-2, subsequently showing a decreasing trend for Ni-MOF-Fe-3 ($\eta_{10} = 294 \text{ mV}$, Tafel slope = 52.7 mV dec^{-1}). Therefore, the Ni-MOF-Fe-2 was synthesized with the optimal Fe ion concentration of 0.02 M .

Additionally, long-term durability is another important criterion for assessing large-scale applications for catalysts. To this end, successive CV scanning tests with a fast speed of 500 mV s^{-1} were exerted on Ni-MOF-Fe-2, and the LSV curve after 5000 cycles overlapped with the initial one (Fig. 3g), demonstrating its outstanding durability. Moreover, the multi-potential process and long-term stability response of the

material were tested. The results in Fig. 3h showed a rapid and recoverable potential response at different current densities, demonstrating the as-prepared sample featured robust mechanical stabilities and excellent electrical conductivity. Fig. S11 (ESI[†]) shows the result of a 30 h chronopotentiometry test, which confirmed the outstanding catalytic stability of Ni-MOF-Fe-2.

To get a deeper insight into possible reasons for the remarkable OER performance of Ni-MOF-Fe-2, data from electrochemical impedance spectroscopy (EIS) were further collected and studied. According to previous reports, EIS is a potential method to investigate the dynamics of electrocatalytic reactions and the behavior of the electrode/electrolyte boundary.^{49,62} Therefore, *Operando* EIS measurements of these synthesized catalysts were carried out. As shown in Fig. 4a, Nyquist plots of the 4 samples exhibited semi-circles with different diameters. Among them, Ni-MOF-Fe-2 possessed the smallest semi-circle, illustrating its fastest electron transfer efficiency. The fitting results in Fig. S12 and Table S3 (ESI[†]) revealed that the charge transfer resistances

(R_{ct}) of these catalysts were 460.23 Ω (Ni MOF), 60.14 Ω (Ni-MOF-Fe-1), 8.31 Ω (Ni-MOF-Fe-2), and 33.39 Ω (Ni-MOF-Fe-3), indicating Fe ion etching can significantly decrease R_{ct} and improve the electrical conductivity of catalysts. In addition, Bode plots of Ni MOF and Ni-MOF-Fe-2 at different applied potentials (from 1.495 V to 1.595 V vs. RHE) were conducted to simulate *Operando* EIS tests, as shown in Fig. 4b and c, and the peak shifted to the high-frequency region with the increasing potential, and this process occurred in the lower frequency region for Ni-MOF-Fe-2, which was related to faster deprotonation of the *OOH intermediate.^{63,64} In Fig. 4d, the phase angle of Ni-MOF-Fe-2 was much smaller than that of Ni-MOF at each potential, indicating more electrons participate in the OER process.⁶⁵ This conclusion was validated by the *Operando* EIS tests of different Fe ion concentrations (Fig. S13, ESI[†]) again. Moreover, we estimated the electrochemical surface area (ECSA) by measuring the double-layer capacitance (C_{dl}) of the catalysts. Fig. S14 (ESI[†]) showed the C_{dl} value of Ni-MOF-Fe-2 was the largest one, demonstrating that

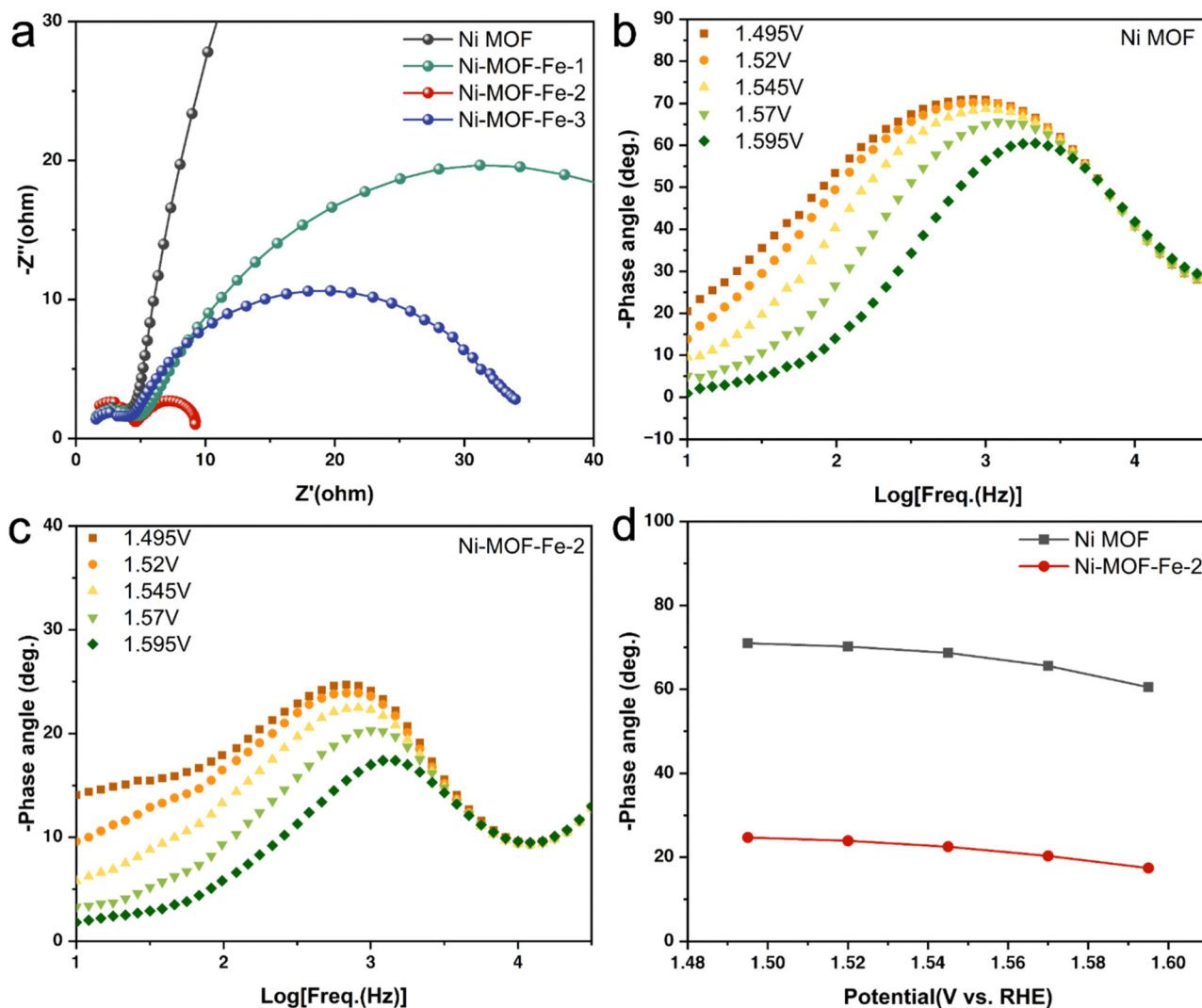


Fig. 4 (a) Nyquist plots of the catalysts at the potential of 1.52 V (vs. RHE). Bode plots of (b) Ni MOF and (c) Ni-MOF-Fe-2 obtained under different potentials. (d) Potential dependence of the phase angle of Ni MOF and Ni-MOF-Fe-2.

the optimal etching concentration yielded the maximum active area and the most active sites for the catalyst, which is well consistent with the OER performance and EIS results described previously.³⁹

Finally, characterization of post-testing Ni-MOF-Fe-2 was conducted. SEM images in Fig. S15 (ESI[†]) showed the post-test sample retained the bulk structure, and its surface got rougher with few spherical amorphous structures left, which may be caused by the surface reconstruction during the OER activation process.⁶¹ XRD pattern in Fig. S16 (ESI[†]) indicated the etched MOF was transformed to nickel oxyhydroxide (PDF#22-0444), which was reasonable in alkaline solution. XPS results in Fig. S17 and S18 (ESI[†]) showed that the ratio of Ni³⁺/Ni²⁺ increased after OER, which was caused by the formation of NiOOH.^{66,67} Additionally, the lattice oxygen centered at about 531 eV evidently increased, confirming the proliferated oxyhydroxide, and it is regarded as the active species in many research works.^{68,69}

Conclusions

In this paper, we designed Fe ion etched nickel-based MOF catalysts to explore their OER performances. As a result, the most outstanding Ni-MOF-Fe-2 reached an overpotential of 269 mV at 10 mA cm⁻² with a small Tafel slope of 47.1 mV dec⁻¹, and it exhibited pretty good durability of 5000 cycles CV test and 30 h chronopotentiometry test. We found that appropriate Fe ion etching modulated the crystallinity and pore structure, and most importantly, the guest element doping formed many dislocations and stacking faults between the lattice planes, becoming active sites and accelerating the kinetics during the OER process. The Ni-MOF-Fe-2 outperformed IrO₂ and inspired a rational design of advanced catalysts. The etching method toward Ni MOFs in this work and thus harvesting their advanced electrochemical performances may create a new path for fabricating other excellent MOF-based catalysts.

Conflicts of interest

There are no conflicts to declare.

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