



Cite this: *Chem. Commun.*, 2020, 56, 9632

Received 7th June 2020,
Accepted 15th July 2020

DOI: 10.1039/d0cc03854g

rsc.li/chemcomm

Control of local flexibility towards *p*-xylene sieving in Hofmann-type porous coordination polymers†

Mohana Shivanna,^{id a} Ken-ichi Otake,^{id a} Jia-Jia Zheng,^{id ab} Shigeyoshi Sakaki^{id b} and Susumu Kitagawa^{id *a}

Adsorption-based xylene isomer separation is more energy efficient than conventional processes. Herein, three isostructural Hofmann-type porous coordination polymers (PCPs), {M(Pz)[Ni(CN)₄]_n} (M = Fe, FePzNi, Co, CoPzNi, and Ni, NiPzNi; Pz = pyrazine) were synthesized and shown to exhibit coordination-dependent lability for the selectivity toward *p*-xylene over *m*- and *o*-xylene.

The separation and purification of important industrial commodities (*e.g.*, gases or solvent vapors) account for 15% of global energy consumption.¹ Among these, xylene isomers (*para*-, *meta*- and *ortho*-xylene abbreviated as *px*, *mx*, and *ox*, respectively) are currently separated from crude oil *via* distillation. This technology involves energy-intensive and time-consuming processes because the similar physicochemical properties of the isomers (boiling point, molecular size, and shape; Table S1, ESI†) make the separation process difficult. However, their separation is crucial for the production of chemical intermediates used to produce many valuable products such as polymer fibers, films, plasticizers, resins, and pigments.² Therefore, other approaches have been employed, such as adsorption, crystallization, complexation, and isomerization.³ In adsorption, physisorbents can effectively resolve the issue of the selective adsorption of one isomer over others because of their pore size and/or pore functionality, which reduces the energy associated with sorbent recycling. In this context, traditional classes of porous solids such as zeolites or silica have been investigated, but are limited to their application as molecular sieves because of the high uptake of all isomers with small difference. For example, FAU zeolite has a selectivity coefficient of only 3–5⁴ and MFI

zeolite has a selectivity of 104 for *px/ox*.⁵ A new subclass of porous materials known as porous coordination polymers (PCPs),^{6,7} or metal organic frameworks (MOFs)⁸ constructed from metal nodes and organic linkers has gained considerable research interest because of the high surface area and extraordinary diverse compositions, making them potential candidates for xylene separation and other applications.^{9,10} The selective uptake of one xylene isomer has been found to be significantly improved compared to that of traditional porous materials.^{11,12} Hofmann clathrates¹³ and Warner complexes¹⁴ are also considered as promising candidates for the separation of xylene isomers.¹⁵ Various strategies have been employed to evaluate their performance, such as their interaction with open metals,¹⁶ pore-size-dependent selectivity in rigid materials,¹¹ breathing induced separation,¹⁷ gate open and close phenomena in switching layered materials,¹⁸ and others.^{19,20} For example, to affirm their flexibility, temperature-dependent tuning of the window aperture of ZIF-8 was shown to demonstrate high selectivity toward *px* over *mx* and *ox*.²¹ Such stimulus-induced transformation of the host network has been exemplified toward guest insertion and removal.^{22,23} Our idea was to utilize pillar rotation to control the diffusion pathway in a pillared layer structure under ambient conditions. Therefore, we chose Hofmann-type three-dimensional PCPs to tune the local flexibility using different metal combinations, that is, tuning the pore window opening toward the recognition of specific xylene isomers. Herein, three isostructural Hofmann-type PCPs with the formula M(Pz)[Ni(CN)_n] (M = Fe, FePzNi, Co, CoPzNi, and Ni, NiPzNi; Pz = pyrazine) were synthesized; the Brunauer–Emmett–Teller (BET) surface areas of these PCPs were in the order FePzNi < CoPzNi < NiPzNi. Interestingly, the observed high *px* adsorption affinity followed the reverse trend (FePzNi > CoPzNi > NiPzNi) and excluded the *mx* and *ox* isomers from the pores (Fig. 1).

Three isostructural PCPs were synthesized using a slight modification of a previously reported procedure²⁴ (a more detailed synthetic procedure is available in the ESI†). The purities

^a Institute for Integrated Cell-Material Sciences (iCeMS), Institute for Advanced Study, Kyoto University (KUIAS), Yoshida Ushinomiya-cho, Sakyo-ku, Kyoto 606-8501, Japan. E-mail: kitagawa@icems.kyoto-u.ac.jp

^b Element Strategy Initiative for Catalyst and Batteries, Kyoto University, Goryo-Ohara 1-30, Nishikyo-ku, Kyoto 615-8245, Japan

† Electronic supplementary information (ESI) available: PXRD, TGA, sorption, and NMR data, and additional experimental and computational details. See DOI: 10.1039/d0cc03854g

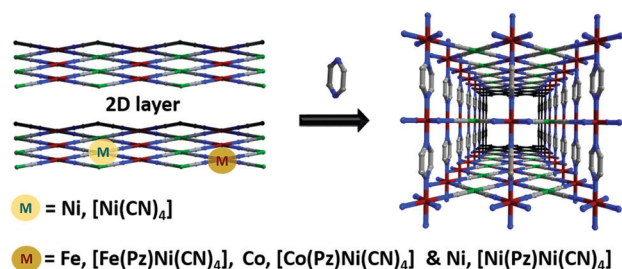


Fig. 1 Schematic representation of the Hofmann-type 2D network pillared by pyrazine used to form a 3D porous coordination polymer. The three isostructural networks are synthesized upon varying the octahedral metal (Fe, Co, Ni) and maintaining the same square planar metal center.

of the as-synthesized phases consisting of FePzNi, CoPzNi, and NiPzNi were confirmed by comparing their powder X-ray diffraction (PXRD) patterns to those calculated from their previously reported crystal structures (Fig. S1–S3, ESI†).^{24–26} Thermogravimetric analysis proved that FePzNi, CoPzNi and NiPzNi were stable up to ~ 250 , ~ 280 and ~ 350 °C (Fig. S4–S6, ESI†). All three PCPs were activated at 120 °C for 12 h under vacuum prior to the sorption measurements. To confirm their porosity, gas sorption isotherm measurements using N₂ at 77 K and CO₂ at 195 K were performed. The three isostructural PCPs exhibit typical type-I profiles toward both N₂ and CO₂. The observed CO₂ and N₂ uptakes were found to be 120 cm³ g^{−1} for FePzNi with a BET surface area of ~ 330 m² g^{−1} (Fig. 2a). In the case of CoPzNi, the CO₂ uptake was slightly lower (140 cm³ g^{−1}) than that observed for N₂ (150 cm³ g^{−1}, BET ~ 430 m² g^{−1}), but higher than FePzNi (Fig. 2b). NiPzNi exhibited a slightly higher uptake for CO₂ (130 cm³ g^{−1}) and N₂ (160 cm³ g^{−1}), but its BET surface area was found to be ~ 500 m² g^{−1} (Fig. 2c).

The analysis of the pore sizes of the crystal structures prompted us to study their xylene vapor sorption isotherms. Interestingly, under vapor pressures of the three xylene isomers, FePzNi exhibits selectivity toward the *px* isomer over the other two, namely *mx* and *ox*. The vapor sorption profile obtained for the *px* isomer was a typical type-I isotherm with an uptake of 40 cm³ g^{−1} at $P/P_0 = 1.0$. However, *mx* (10 cm³ g^{−1}) and *ox* (5 cm³ g^{−1}) showed negligible uptake, which was limited to surface adsorption only (Fig. 3a). This indicates that FePzNi

shows high selectivity toward *px* (6.7 Å), which has a slightly smaller kinetic diameter when compared to *mx* (7.1 Å) and *ox* (7.4 Å).⁴ In contrast, CoPzNi exhibits S-shaped or gate opening²⁷ type adsorption behavior toward the *px* isomer; a slight uptake from 0 to 5 cm³ g^{−1} was observed upon increasing the pressure up to $P/P_0 = 0.18$ (the gate-open pressure), followed by a sudden increase in the uptake to 30 cm³ g^{−1} prior to $P/P_0 = 0.22$, which saturated to 40 cm³ g^{−1} at $P/P_0 = 1.0$. The desorption process follows the adsorption profile and finally reverts to the activated phase, as verified by PXRD (Fig. S2, ESI†). CoPzNi exhibited a low affinity toward the other two isomers with uptakes of 9 cm³ g^{−1} (*mx*) and 4 cm³ g^{−1} (*ox*) (Fig. 3b). NiPzNi exhibits a small non-selective uptake of the three isomers of 7 (*px*), 5 (*mx*), and 3 cm³ g^{−1} (*ox*) (Fig. 3c). When the temperature was increased to 308 K during the vapor sorption of *px*, the gate-open pressure observed for CoPzNi showed a slight increase to $P/P_0 = 0.2$ (Fig. S9, ESI†). FePzNi and NiPzNi exhibit no difference in their sorption performance when compared to those observed at 298 K. The recyclability tests using FePzNi and CoPzNi indicate that their *px* sorption profiles were reproducible with the same uptake maintained over 5 cycles (Fig. S10 and S11, ESI†). Importantly, the adsorption affinity toward *px* follows the reverse trend (FePzNi > CoPzNi > NiPzNi) compared to the trend for the BET surface area (FePzNi < CoPzNi < NiPzNi). To determine the binary mixture separation performance, we conducted vapor phase experiments using 1:1 mixture of xylene isomers. The selectivities were estimated by the integration area ratio of the adsorbed components by the ¹H NMR spectra (Fig. S15–S18; see ESI† for the detail). Interestingly, FePzNi exhibited an exceptional *px* selectivity when exposed to *px/mx* mixture. While upon *px/ox* mixture, FePzNi showed the selectivity of 2. Whereas, CoPzNi showed *px* selectivities of 1.2 and 1.15 over *px/mx* and *px/ox* mixture, respectively. Except for FePzNi towards *px/mx*, these calculated selectivities are slightly higher or comparable to previously reported materials²⁸ (Table S2, ESI†).

To gain insight into mechanism of *px* selective adsorption and the structural transformation, we collected PXRD data by soaking a few milligrams of each PCP in three different vials containing the pure liquid (*px* isomer) for 24 h. The PXRD patterns indicate negligible changes in FePzNi and NiPzNi (Fig. S1 and S3, ESI†).

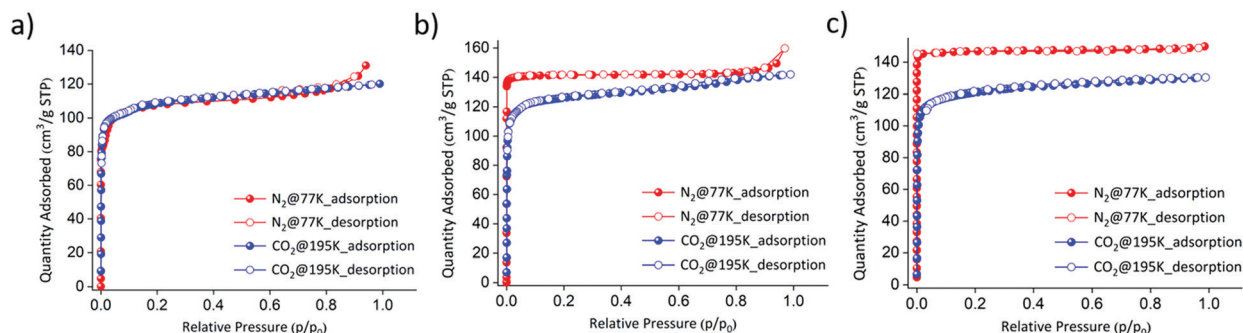


Fig. 2 Gas sorption (blue, CO₂ at 195 K; red, N₂ at 77 K) on three isostructural PCPs: (a) FePzNi (calculated BET surface area using N₂ was found to be ~ 330 m² g^{−1}), (b) CoPzNi (BET surface area using N₂ ~ 430 m² g^{−1}), and (c) NiPzNi (BET surface area using N₂ ~ 500 m² g^{−1}).

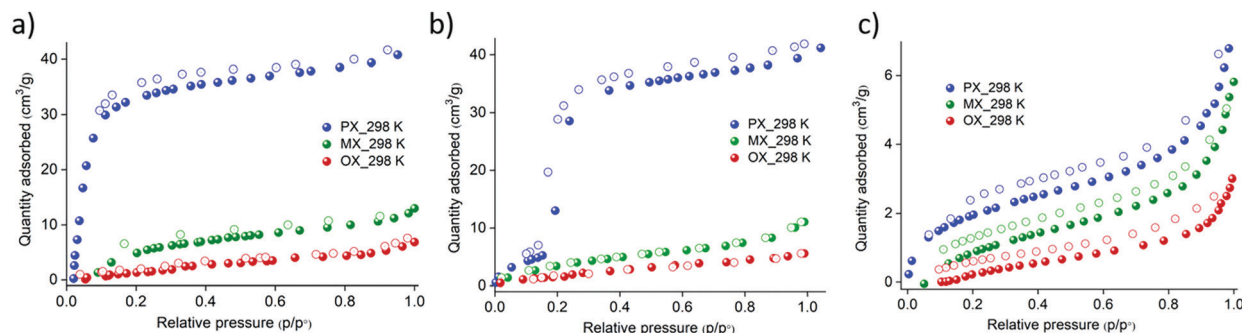


Fig. 3 Pure component xylene isomer (*px*-blue, *mx*-green, and *ox*-red) sorption on the three isostructural PCPs measured at 298 K (a) FePzNi exhibits a typical type-I isotherm for *px* with an uptake of 40 cm³ g⁻¹. (b) CoPzNi exhibits a stepped profile during the sorption of *px* with the same uptake as FePzNi. (c) No gate opening was observed for *px* when using NiPzNi. All three PCPs shown negligible uptake of *mx* and *ox*.

CoNiPz exhibits some shifts in the peaks corresponding to (001) and (002), which are directed toward the pyrazine and 2D metal layers (Fig. S2 and S19, ESI†). This implies that upon *px* adsorption pyrazine undergoes rotation and that the subtle transformation of the 2D layer leads to a stepped profile. We further analyzed the crystal structures in order to gain an insight into the lability of the coordination bonds. In the as-synthesized form of FePzNi, the pyrazine molecules align parallel to the channel along the *b*-axis, while upon removal of the guest molecules (water), the pyrazine molecules are distorted to close the channel window (Fig. 4a). When the pyrazine molecules align parallel to the channel direction, the centroid distance between the two aromatic moieties (C–C... π with $D_{C...C} = 7.257(5)$ Å) is suitable for the *px* isomer when compared to *mx* and *ox*. In the perpendicular direction, the π ... π distance was reduced to 5.1(3) Å. The Fe–Fe distances

along the diagonal and between the 2D layers were found to be $D_{Fe1...Fe3} = 10.262(23)$ Å and $D_{Fe1...Fe4} = 7.256(3)$ Å, respectively (Fig. S20, ESI† and Table 1). The distances between the octahedral center of Fe and the N atom of the cyanide was $D_{Fe...N} = 2.117(44)$ Å and N atom of the pyrazine was $D_{Fe...N} = 2.212(116)$ Å. The square planar metal center (NiCN₄) was aligned with $D_{Ni...C} = 1.864(44)$ Å (Fig. S20, ESI† and Table 1). Most of these bond distances are slightly shorter in CoPzNi and much shorter in NiPzNi structures (Fig. S21 and S22, ESI† and Table 1). Therefore, as shown in Fig. 4a, pyrazine rotation is a key and controlling factor to open the channel gate toward *px* selectivity. The 2D interlayer distance (7.256(3) Å) can allow facile ring rotation in FePzNi, but the slightly reduced distance (7.107(6) Å), which pushes to higher pressures in CoPzNi and much smaller distance (7.024(4) Å) makes it difficult for the NiPzNi structure.

The rotational barrier of the Pz pillar (Fig. 4b) and deformation energy increases in the order FePzNi (2.6 kcal mol⁻¹) < CoPzNi (3.7 kcal mol⁻¹) < NiPzNi (4.7 kcal mol⁻¹) and FePzNi (2.5 kcal mol⁻¹) < CoPzNi (2.7 kcal mol⁻¹) < NiPzNi (3.2 kcal mol⁻¹) (Table S3, ESI†) which is consistent with these geometrical features and indicates that this steric effect plays an important role in the rotation of the pyrazine molecules in these PCPs, as discussed in our previous work.²³ In addition, the large space in FePzNi makes it the best for *px* adsorption among the three Hofmann-type PCPs studied, where the binding energy (see ESI† for the definition of the binding energy and computational details) of *px* decreases in the order of FePzNi (–18.3 kcal mol⁻¹) > CoPzNi (–16.1 kcal mol⁻¹) > NiPzNi (–13.8 kcal mol⁻¹). While interaction energy between *px* and PCPs follows same trend, FePzNi (–20.8 kcal mol⁻¹) > CoPzNi (–18.8 kcal mol⁻¹) > NiPzNi (–17.0 kcal mol⁻¹) (Table S3, ESI†), indicating the largest affinity of FePzNi toward *px*. These results suggest that an appropriate distance between the pyrazine rings in FePzNi was beneficial for *px* adsorption both kinetically and thermodynamically.

In conclusion, three isostructural Hofmann-type PCPs exhibit control over pillar rotation to recognize specific xylene isomers. When FePzNi is exposed to xylene vapor, the pyrazine rotation showed a high affinity toward the *px* isomer with a typical type-I isotherm. In the case of the CoPzNi structure upon *px* sorption, pyrazine rotation required a slightly higher

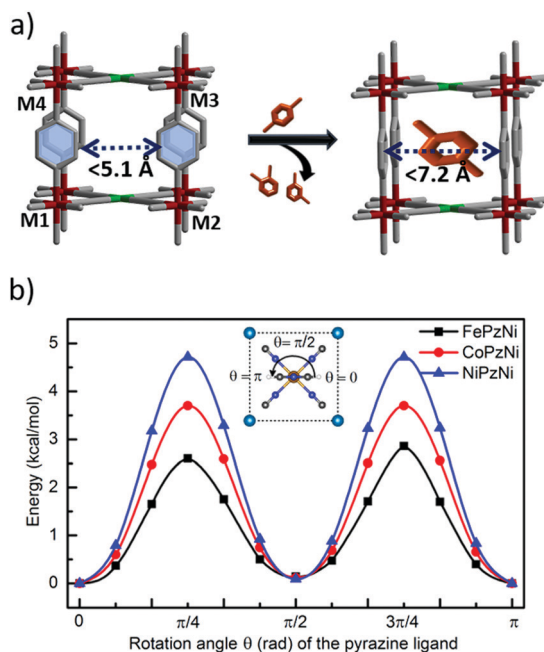


Fig. 4 (a) A schematic representation of the pillar rotation toward the *px* isomer. (b) The potential energy surface for the rotation of the pyrazine pillars in FePzNi, CoPzNi, and NiPzNi.

Table 1 Structural parameters for MPzNi (M = Fe, Co, Ni)

	$D_{M \cdots M}$	$D_{M \cdots N(CN)}$	$D_{M \cdots N(Pz)}$	$D_{Ni \cdots C(CN_4)}$
FePzNi	$D_{Fe1 \cdots Fe3} = 10.262(23)$, $D_{Fe1 \cdots Fe4} = 7.256(3)$ Å	$D_{Fe \cdots N} = 2.117(44)$ Å	$D_{Fe \cdots N} = 2.212(116)$ Å	$1.864(44)$ Å
CoPzNi	$D_{Co1 \cdots Co3} = 10.131(6)$, $D_{Co1 \cdots Co4} = 7.107(6)$ Å	$D_{Co \cdots N} = 2.089(36)$ Å	$D_{Co \cdots N} = 2.167(64)$ Å	$1.858(43)$ Å
NiPzNi	$D_{Ni1 \cdots Ni3} = 10.027(4)$, $D_{Ni1 \cdots Ni4} = 7.024(4)$ Å	$D_{Ni \cdots N} = 2.160(215)$ Å	$D_{Ni \cdots N} = 2.037(7)$ Å	$1.825(209)$ Å

energy and led to a stepped or gate opening isotherm with no considerable hysteresis. The high uptake and selectivity due to the pore size is a perfect fit for the *px* isomer after pillar rotation. However, NiPzNi behaves as a highly rigid material and does not induce any structural transformation of the pillar under a partial pressure of any of the three xylene isomers. The results indicate that coordination is a key factor in controlling these subtle transformations, which can allow specific guest molecules into the pores. Therefore, this work demonstrates that the construction of a framework with desirable metal coordination plays an important role in the separation of xylene isomers. Moreover, their ease of synthesis from common and inexpensive starting chemicals may lead to the design of potential candidates that can be applied for industrial-related molecular recognition. Separation studies of C₂ gas mixtures with NiPzNi are in progress.

Authors thank the financial support of KAKENHI, Grant-in-Aid for Scientific Research (S) (JP18H05262), and Early-Career Scientists (JP19K15584) from the Japan Society of the Promotion of Science (JSPS). Synchrotron XRD measurements were supported by the Japan Synchrotron Radiation Research Institute (JASRI) (Proposal No. 2019A1600, 2019B1554, 2020A1496). We thank Dr Kawaguchi of JASRI and Prof. Kubota of Osaka Prefecture University for their experimental help at SPring-8. NMR facility was used at the iCeMS Analysis Center, KUIAS.

Conflicts of interest

There are no conflicts to declare.

Notes and references

- D. S. Sholl and R. P. Lively, *Nature*, 2016, **532**, 435–437.
- N. Sun, S.-Q. Wang, R. Zou, W.-G. Cui, A. Zhang, T. Zhang, Q. Li, Z.-Z. Zhuang, Y.-H. Zhang, J. Xu, M. J. Zaworotko and X.-H. Bu, *Chem. Sci.*, 2019, **10**, 8850–8854.
- M. du Plessis, V. I. Nikolayenko and L. J. Barbour, *J. Am. Chem. Soc.*, 2020, **142**, 4529–4533.
- Y. Yang, P. Bai and X. Guo, *Ind. Eng. Chem. Res.*, 2017, **56**, 14725–14753.
- G.-Q. Guo, H. Chen and Y.-C. Long, *Microporous Mesoporous Mater.*, 2000, **39**, 149–161.
- S. Kitagawa, R. Kitaura and S. Noro, *Angew. Chem. Int. Ed.*, 2004, **43**, 2334–2375.
- S. R. Batten, S. M. Neville and D. R. Turner, *Coordination polymers: design, analysis and application introduction*, RSC Publishing, Cambridge, UK, 2009.
- M. Schroder, Functional Metal-Organic Frameworks: Gas Storage, Separation and Catalysis, in *Topics in Current Chemistry*, ed. M. Schroder, 2010, vol. 293, pp. 1–262.
- J. Lee, O. K. Farha, J. Roberts, K. A. Scheidt, S. T. Nguyen and J. T. Hupp, *Chem. Soc. Rev.*, 2009, **38**, 1450–1459.
- S. Horike, S. Shimomura and S. Kitagawa, *Nat. Chem.*, 2009, **1**, 695–704.
- B. Saccoccia, A. M. Bohnsack, N. W. Waggoner, K. H. Cho, J. S. Lee, D.-Y. Hong, V. M. Lynch, J.-S. Chang and S. M. Humphrey, *Angew. Chem. Int. Ed.*, 2015, **54**, 5394–5398.
- J. E. Warren, C. G. Perkins, K. E. Jelfs, P. Boldrin, P. A. Chater, G. J. Miller, T. D. Manning, M. E. Briggs, K. C. Stylianou, J. B. Claridge and M. J. Rosseinsky, *Angew. Chem. Int. Ed.*, 2014, **53**, 4592–4596.
- H. M. Powell, *J. Am. Chem. Soc.*, 1948, 61–73.
- W. D. Schaeffer, W. S. Dorsey, D. A. Skinner and C. G. Christian, *J. Am. Chem. Soc.*, 1957, **79**, 5870–5876.
- M. Lusi and L. J. Barbour, *Angew. Chem. Int. Ed.*, 2012, **51**, 3928–3931.
- M. I. Gonzalez, M. T. Kapelewski, E. D. Bloch, P. J. Milner, D. A. Reed, M. R. Hudson, J. A. Mason, G. Barin, C. M. Brown and J. R. Long, *J. Am. Chem. Soc.*, 2018, **140**, 3412–3422.
- V. Finsy, C. E. A. Kirschhock, G. Vedts, M. Maes, L. Alaerts, D. E. De Vos, G. V. Baron and J. F. M. Denayer, *Chem. – Eur. J.*, 2009, **15**, 7724–7731.
- S.-Q. Wang, S. Mukherjee, E. Patyk-Kaźmierczak, S. Darwish, A. Bajpai, Q.-Y. Yang and M. J. Zaworotko, *Angew. Chem. Int. Ed.*, 2019, **58**, 6630–6634.
- F. Vermoortele, M. Maes, P. Z. Moghadam, M. J. Lennox, F. Ragon, M. Boulhout, S. Biswas, K. G. M. Laurier, I. Beurroies, R. Denoyel, M. Roeflaers, N. Stock, T. Düren, C. Serre and D. E. De Vos, *J. Am. Chem. Soc.*, 2011, **133**, 18526–18529.
- V. Finsy, H. Verelst, L. Alaerts, D. De Vos, P. A. Jacobs, G. V. Baron and J. F. M. Denayer, *J. Am. Chem. Soc.*, 2008, **130**, 7110–7118.
- D. M. Polyukhov, A. S. Poryvaev, S. A. Gromilov and M. V. Fedin, *Nano Lett.*, 2019, **19**, 6506–6510.
- M. Ohba, K. Yoneda, G. Agustí, M. C. Muñoz, A. B. Gaspar, J. A. Real, M. Yamasaki, H. Ando, Y. Nakao, S. Sakaki and S. Kitagawa, *Angew. Chem. Int. Ed.*, 2009, **48**, 4767–4771.
- H. Ando, Y. Nakao, H. Sato, M. Ohba, S. Kitagawa and S. Sakaki, *Chem. Phys. Lett.*, 2011, **511**, 399–404.
- P. D. Southon, L. Liu, E. A. Fellows, D. J. Price, G. J. Halder, K. W. Chapman, B. Moubaraki, K. S. Murray, J.-F. Létard and C. J. Kepert, *J. Am. Chem. Soc.*, 2009, **131**, 10998–11009.
- J. Rodríguez-Hernández, A. A. Lemus-Santana, J. Ortiz-López, S. Jiménez-Sandoval and E. Reguera, *J. Solid State Chem.*, 2010, **183**, 105–113.
- J. Pei, K. Shao, J.-X. Wang, H.-M. Wen, Y. Yang, Y. Cui, R. Krishna, B. Li and G. Qian, *Adv. Mater.*, 2020, 1908275.
- Q. Y. Yang, P. Lama, S. Sen, M. Lusi, K. J. Chen, W. Y. Gao, M. Shivanna, T. Pham, N. Hosono, S. Kusaka, J. J. T. Perry, S. Ma, B. Space, L. J. Barbour, S. Kitagawa and M. J. Zaworotko, *Angew. Chem. Int. Ed.*, 2018, **57**, 5684–5689.
- L. Alaerts, C. E. A. Kirschhock, M. Maes, M. A. van der Veen, V. Finsy, A. Depla, J. A. Martens, G. V. Baron, P. A. Jacobs, J. F. M. Denayer and D. E. De Vos, *Angew. Chem. Int. Ed.*, 2007, **46**, 4293–4297.