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Solvation and interfacial chemistry in ionic liquid based electrolytes toward rechargeable lithium-metal batteries

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Rechargeable lithium metal batteries (LMBs) are a highly promising technology for high-energy-density storage systems due to the low electrochemical potential and high theoretical capacity of the lithium metal anode. The electrolyte plays a pivotal role among the critical components of LMBs. However, traditional organic electrolytes pose significant safety risks and shorten the battery life due to their electrochemical instability, volatility, and flammability. Alternatively, ionic liquids (ILs), composed of anions and cations, are room-temperature molten salts characterized by ultra-low volatility, high ionic conductivity, excellent thermal stability, low flammability, and wide electrochemical windows. Based on these properties, liquid IL electrolytes (ILEs) and polymeric IL electrolytes (PILEs) have shown immense potential in enhancing battery cycle stability, energy density, lifespan, and safety. This review aims to comprehensively explore and summarize recent applications of ILEs and PILEs in LMBs, including their use as liquid and solid-state electrolytes, as well as ILs serving as film-forming additives, interfacial wetting agents, and pretreatment reagents. Additionally, the review delves into the solvation structures of Li^+ ions within different IL-based electrolytes and the resulting interfacial chemical characteristics. Finally, based on literature reports and our previous work, we identify current challenges and propose solutions for the future application of IL-based electrolytes in LMBs.

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1 Introduction

Thirty years ago, Sony commercialized the first lithium-ion batteries (LIBs) by using $LiCoO_2$ (LCO) as the cathode, graphite (Gr) as the anode, and an ethylene carbonate (EC)-based electrolyte.[1] With continuous technological advancements and iterations, LIBs have been widely used in modern electronic devices such as smartphones and laptops. However, the energy density of current LIBs is approaching its theoretical value due to the limited specific capacity of the Gr anode (374 mA h g^{-1}), which cannot satisfy the market requirement.[2–4] Alternatively, lithium metal batteries (LMBs), which utilize lithium metal as the anode (LMA), can theoretically provide a much higher specific capacity of up to 3860 mA h g^{-1}, far exceeding that of the traditional Gr anode.[5–7] Additionally, LMBs can be paired with high-capacity cathode materials to further boost the energy density, including nickel-rich, layered transition metal oxides ($LiNi_xMn_yCo_zO_2$, NCM), cobalt-free $Li_{1.2}Ni_{0.2}Mn_{0.6}O_2$ (LNM), spinel $LiNi_{0.5}Mn_{1.5}O_4$ (LNMO), and sulfur (S).[8–11]

However, traditional carbonate-based and ether-based electrolytes undergo uncontrollable side reactions with LMA, leading to anode pulverization and electrolyte depletion.[12] Moreover, the electrode/electrolyte interfaces (EEIs), including the solid electrolyte interphase (SEI) and the cathode electrolyte interphase (CEI), are chemically unstable and mechanically fragile.[13,14] The organic-rich SEI fails to suppress dendrite growth and continuous Li consumption, resulting in poor cycling stability.[15] Moreover, the unstable CEI cannot effectively inhibit the catalytic decomposition of the electrolyte by the cathode material. For example, high-nickel NCM cathodes undergo irreversible phase transitions and metal ion dissolution during cycling, limiting the application of high-voltage LMBs.[16] Among numerous efforts in the literature, innovative electrolyte designs offer a cost-effective and convenient solution for the current development of LMBs.[17–19]

Ionic liquids (ILs) are salts with melting points below 100 °C, composed of organic cations and organic or inorganic anions.[20] Paul Walden first discovered and reported ethylammonium nitrate in 1914, marking the beginning of a surge in IL research.[21] In recent years, the number of studies on ILs as electrolyte materials (ILEs) has significantly increased, as evidenced by the growing number of related articles published in academic journals (Fig. 1a), reflecting researchers' strong interest in this field. In general, as shown in Fig. 1b, several key parameters should be considered using ILs as electrolytes for rechargeable LMBs. They have low melting point, low viscosity, wide electrochemical window, low volatility and non-flammability, and high ionic conductivity, which are essential for advancing the cycle performance for LMBs.

Based on molecular characteristics, ILs can be divided into small molecule zwitterionic ILs and poly(ionic liquids) (PILs).[22] PILs are polymeric materials formed by the polymerization of IL monomers,[23] combining the properties of polymers and ILs, typically exhibiting higher mechanical strength and stability while retaining the excellent ionic conductivity and chemical stability. The structure of PILs usually includes a monomer with positive and negative ions and a polymer backbone, which can further be categorized into polycation ILs (PCILs), polyanion ILs (PAILs), and poly(zwitterion) ILs (PZILs).[24] In addition, by adjusting the type of monomers and polymerization methods, PILs with different physicochemical properties can be designed

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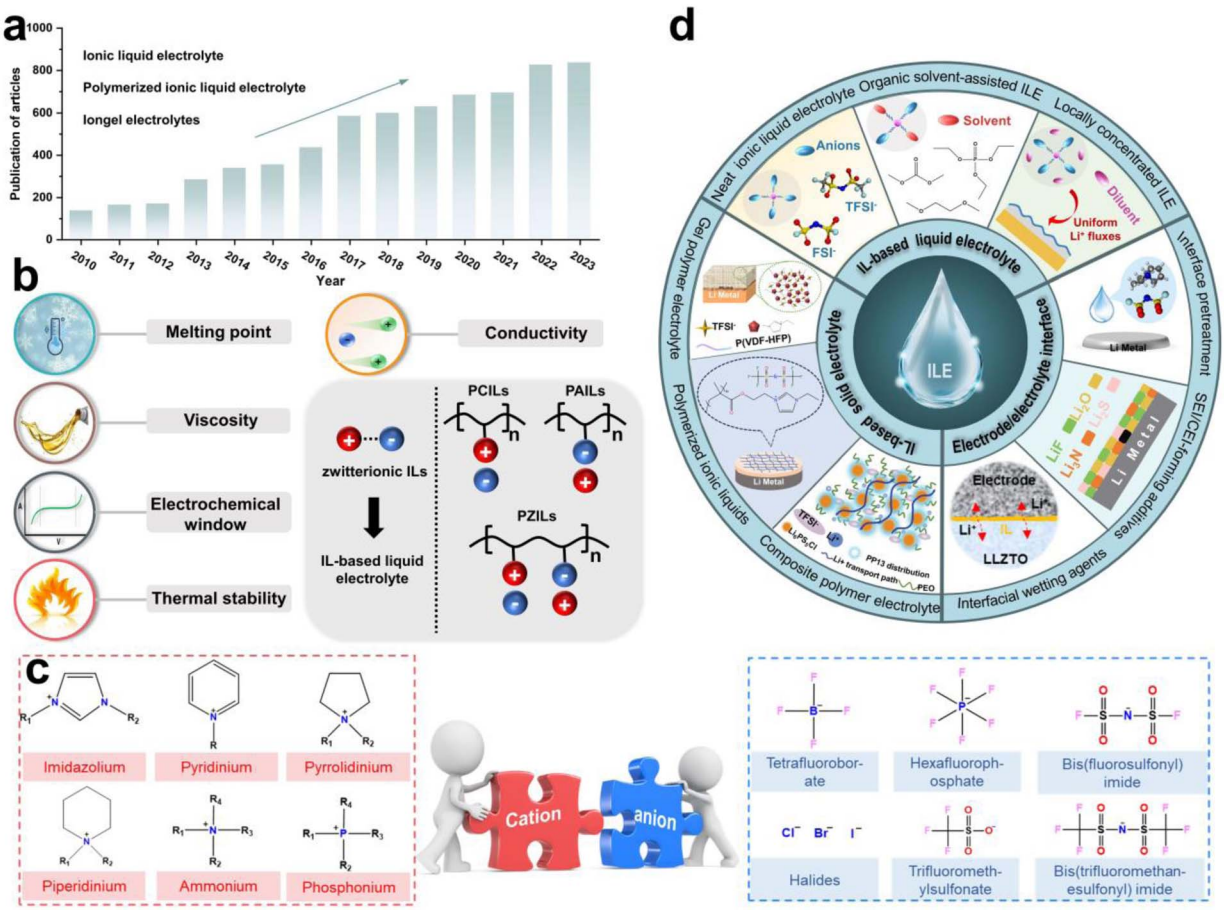

Fig. 1 (a) Number of publications of ionic liquids applied in batteries. Data were obtained from Web of Science on the 30th May, 2024, *via* keywords such as ionic liquid electrolyte, polymerized ionic liquid electrolyte and iongel electrolyte. (b) The key parameters for evaluating ionic liquids as electrolytes. (c) Chemical structures of various ILs. (d) Summary of ionic liquid application.

for solid-state LMBs. Fig. 1c presents common organic cations and inorganic or organic anions, including imidazolium, pyridinium, tetrafluoroborate, hexafluorophosphate and so on. Several reviews have summarized the application of ILs in batteries. For instance, Wu *et al.* reviewed the progress of IL-based electrolytes in non-aqueous and aqueous metal batteries, and Zhou *et al.* reported on the composition, classification, and application of common IL electrolytes in different valence state ion batteries.[25,26] However, a systematic review focusing on the application of ILs in LMBs from the perspective of electrolyte solvation chemistry and the interfacial chemistry formed by ILs is still lacking.

Herein, based on our previous work and the recent reported literature on IL-based electrolytes in LMBs, we have systematically summarized the IL-based liquid and solid electrolytes, as well as the use of ILs as additives or wetting agents for EEI treatment (Fig. 1d). The solvation structure of Li^+ ions and the unique interfacial chemical components formed are discussed in detail. We propose that the application of ILs and PILs in LMBs is still in its early stages, with a lack of fundamental understanding of the solvation chemistry and advanced *in situ* characterization techniques for interfacial chemical components. Furthermore, there remains significant potential for the molecular design of ILs. For example, machine learning and artificial intelligence (AI) big data screening can efficiently design target IL molecules. Finally, we propose a novel concept for the application of PILs in solid electrolytes-the solidification of electrodes, which expands the imaginative scope for designing solid-state electrolytes and non-flammable liquid ILEs. This timely and critical review provides an overview of the latest design strategies for ILs in LMB applications, offering insights for the design of next-generation ILEs based on solvation chemistry and interfacial chemistry modulation.

2 IL-based liquid electrolytes

Traditional carbonate- or ether-based electrolytes suffer from uncontrolled side reactions with electrodes, forming chemically unstable SEI and CEI, which in turn results in dendrite growth, dead Li, and continuous Li consumption.[27,28] Alternatively, ILs based on cations such as imidazolium or pyrrolidinium exhibit excellent film-forming ability and a wide electrochemical window, making them suitable for LMBs operating under high voltages or other special environmental conditions.[29] ILEs also possess low volatility and non-flammability, which can significantly enhance the battery safety.[30] Compared to IL-based solid-

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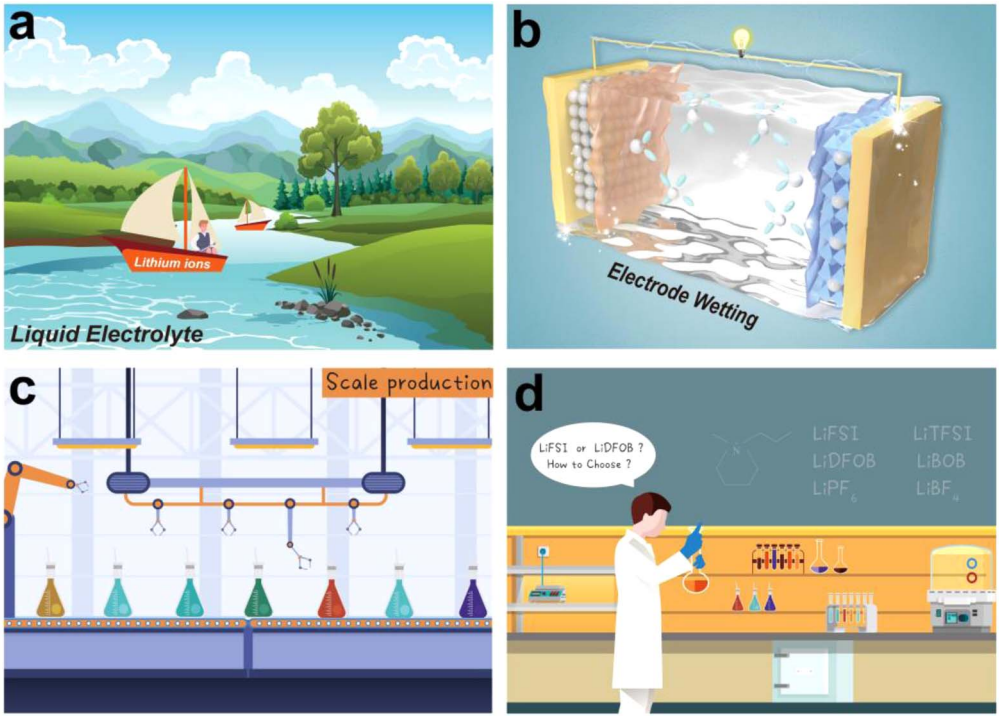

Fig. 2 The advantages of liquid ILEs. (a) Schematic of rapid Li^+ ion transport in liquid ILEs. The river represents the liquid IL, and the sailboats represent Li^+ ions. Schematic illustrations of liquid electrolytes (b) wetting electrodes, (c) large-scale production, and (d) design flexibility.

state electrolytes, IL-based liquid electrolytes have numerous advantages in LMBs: (i) high ionic conductivity to facilitate more efficient Li^+ ion transport, and enhance the rate performance of the battery (Fig. 2a); (ii) better wettability to reduce interfacial resistance, and improve overall battery performance (Fig. 2b); (iii) well-established preparation to enable easy large-scale production (Fig. 2c); (iv) significant flexibility in design and application to tune their performance (Fig. 2d). In summary, based on the types of electrolyte solvents, liquid ILEs can be classified into neat ILEs, organic solvent-assisted ILEs, and locally concentrated ILEs.

2.1 Neat ionic liquid electrolytes

The advancement of neat ILEs can be categorized into two key areas: the regulation of solvation structures and the design of organic cations. A common approach to solvation structure regulation involves using dual-anion ILEs. In such systems, the competitive interactions between $TFSI^-$ and FSI^- anions slow the reaction kinetics of FSI^-, leading to the formation of a thin and dense SEI layer. Compared to single-anion ILEs, dual-anion ILEs show superior compatibility with lithium metal, resulting in more uniform, compact, and dendrite-free Li deposition. In high-concentration ILEs, the proportion of contact ion pairs (CIPs) and aggregates (AGGs) in the solvation structure increases, creating an anion-rich environment. This enhances the reduction of anions, increasing the LiF content at the interface and stabilizing both the anode and cathode. However, the redox activity of organic cations can lead to higher organic content at the EEI, reducing the interface stability and increasing the impedance. To address this, redesigning organic cations can significantly improve the performance of neat ILEs by introducing ether oxygen atoms into side chains, extending the chain length, and fluorinating the cations. This section provides a comprehensive overview of the mechanisms behind the design strategies for neat ILEs.

Wu *et al.* reported an ionic liquid electrolyte containing both FSI^- and $TFSI^-$ anions, denoted as $0.8Pyr_{14}FSI$–$0.2LiTFSI$ (Fig. 3a).[31] The synergistic effect of FSI^- and $TFSI^-$ enabled NCM88 ($LiNi_{0.88}Co_{0.09}Mn_{0.03}O_2$) to exhibit excellent long-cycle stability. In this electrolyte, Li/NCM88 batteries showed 88% capacity retention after 1000 cycles with almost no voltage decay, and the average coulombic efficiency exceeded 99.94%. From the cross-section images of the cycled cathode, commercial electrolytes led to various cracks and particle fractures in electrodes at high states of charge (SOC) due to stress from expansion and the H2–H3 phase transition after repeated charge–discharge cycles. In contrast, the NCM88 cathode cycled in the ILE showed no severe cracking indicating that the CEI film is more stable. Neat ILEs can also form a stable SEI on LMA. Using reactive molecular dynamics simulations (MD), the atomic details of SEI formation at the LMA and ILE interface can be observed. Yang *et al.* developed an atomic model of SEI formation at the interface between LMA and the [BMIM][TFSI] (bis(trifluoromethylsulfonyl)-imide/1-butyl-3-methyl-imidazolium) ILE (Fig. 3b).[32] Reactive force field (ReaxFF) molecular dynamics simulations revealed significant differences in the decomposition rates and extent of $TFSI^-$ anions and $BMIM^+$ cations, with $TFSI^-$ decomposition products being predominant. At 300 K, the SEI exhibited a bilayer structure

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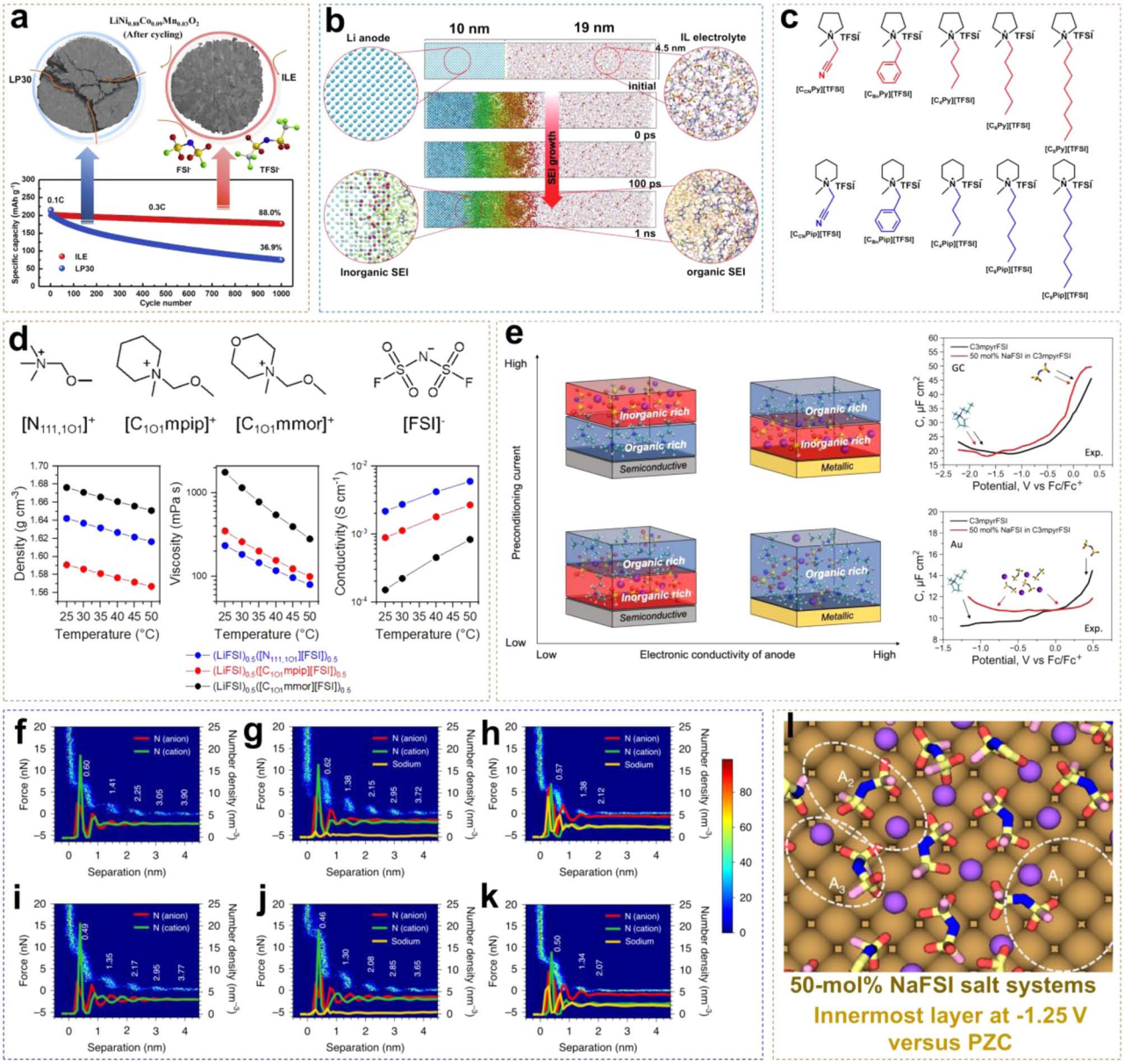

Fig. 3 (a) Long-term cycling performance of an exemplary Li/NCM88 cell using ILE and LP30 electrolyte. Reprinted from ref. 31 with permission. Copyright 2021, Elsevier. (b) The formation of the SEI layer from the ReaxFF MD simulation at 300 K using a [BMIM] [TFSI] IL electrolyte. Reprinted from ref. 32 with permission. Copyright 2022, Wiley. (c) Chemical structures of pyrrolidinium- and piperidinium-based ILs. Reprinted from ref. 33 with permission. Copyright 2020, American Chemical Society. (d) Chemical structures and corresponding density, viscosity, and conductivity of different ether-functionalized ILEs. Reprinted from ref. 34 with permission. Copyright 2023, Elsevier. (e) Schematic relationships between the SEI composition, applied preconditioning current, and dielectric nature of the anode material. Reprinted from ref. 35 with permission. Copyright 2023, The Royal Society of Chemistry. AFM force–distance two-dimensional histograms and ion number density profiles of FSI^-, C_3mpyr^+ and Na^+ from MD simulations at OCP (AFM) and PZC (MD) (f–h) and OCP −0.5 V (AFM) and −0.5 V *versus* PZC (MD) (i–k) for C3mpyrFSI IL with different NaFSI concentrations of 0 mol% (f, and i), 10 mol% (g, and j), and 50 mol% (h, and k). (l) Typical Na–FSI coordination complexes in the innermost layer at −1.25 V *versus* PZC in the 50 mol% salt systems. Reprinted from ref. 36 with permission. Copyright 2020, Springer Nature.

with a 2.5 nm ordered inorganic layer near the LMA and a 7.5 nm porous organic layer near the ILE. The excellent electrochemical stability of ILEs in batteries is attributed not only to the formation of a stable EEI but also to their intrinsically wide voltage window. Qi *et al.* systematically synthesized piperidinium- and pyrrolidinium-based ILs with various substituents, including cyanomethyl, benzyl, butyl, hexyl, and octyl, and investigated their electrochemical stability.[33] As shown in Fig. 3c, introducing electron-donating substituents into the side chains can enhance the oxidative stability of ILs. For example, $[C_6Py][TFSI]$ has a decomposition voltage of 5.2 V, higher than that of $[C_{CN}Py][TFSI]$ (3.81 V). The order with respect to decomposition voltage for piperidinium-based ILs is similar to that of pyrrolidinium-based ILs, with $[C_6Pip][TFSI]$ exhibiting the highest decomposition voltage of 5.09 V. When the substituent changes from a hexyl group to an octyl group, the

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change of voltage stability is negligible due to the reduced electron-donating ability with the increased length of the carbon chain. Unlike the work contributed by Qi *et al.*, Warrington *et al.* functionalized the side chains with ether groups to enhance the fluidity of ILs and reduce Li^+–anion interactions in highly concentrated ionic liquid electrolytes (HCILEs), facilitating faster Li^+ ion transport.[34] As shown in Fig. 3d, three novel ILEs were compared, specifically $[N_{111,1O1}][FSI]$ (1-methoxymethyl-1,1,1-tri-methylammonium), $[C_{1O1}mpip][FSI]$ (*N*-methoxymethyl-*N*-methylpiperidinium), and $[C_{1O1}mmor][FSI]$ (*N*-methoxymethyl-*N*-methylmorpholinium). It was found that $(LiFSI)_{0.5}([N_{111,1O1}][FSI])_{0.5}$ exhibited the lowest viscosity and the highest ionic conductivity. This work demonstrates the flexibility in the design and synthesis of ILEs.

Forsyth *et al.* found that the composition of the EEI in ILEs is significantly influenced by the dielectric properties of the electrode material.[35] As shown in Fig. 3e, ILEs form an organic–inorganic double layer SEI on semiconductor electrodes. At low current densities, the inner and outer layers are composed of inorganic and organic components, respectively; at high current densities, this order is reversed. This is attributed to the increased surface negative charge on the semiconductor electrode with rising current, which increases the number of IL cations. The metallic electrode surface, characterized by high electron polarizability and strong dispersion forces, requires higher current densities to form an anion-derived interface. Differential capacitance measurements, obtained by recording electrochemical impedance spectra (EIS) at various applied potentials in the non-faradaic region, reveal that on semiconductive glassy carbon (GC) electrodes, the *C–E* (differential capacitance–electrical potential) curve of pure C_3mpyrFSI exhibits a descending shape from positive to negative potential regions. This behavior arises from the tighter packing of $[FSI]^-$ anions compared to cations at the interface, resulting in increased charge accumulation and higher capacitance. However, the shape of the *C–E* curve on Au electrodes is significantly influenced by the dominance of the Na_xFSI_y species. This also indicates a substantial difference in the composition of the electric double layer (EDL) across different electrode surfaces. Rakov *et al.* investigated the nanostructure of IL electrode surfaces at NaFSI concentrations and open circuit potential (OCP) using atomic force microscopy (AFM) and MD simulations.[36] They found that at a distance of 4.5 nm from the electrode, there exists multiple nanostructured layers that decay into the bulk phase (Fig. 3f). By comparison, adding 10 mol% NaFSI salt enhances the rupture force required to disrupt each layer due to increased ion–ion association (Fig. 3g). In a system with 50 mol% NaFSI, the force significantly decreases after only two to three steps, indicating a change in the physical dimensions of interfacial ion packing in the HCILE system (Fig. 3h). When a potential of −0.5 V is applied to the working electrode, the impact on the number of ion layers becomes negligible (Fig. 3i–k). The ion number density profiles obtained from MD simulations align with AFM measurements, showing the formation of molten salt-like $Na_x(FSI)_y$ ion aggregates at the electrode interface in HCILE systems (Fig. 3l). Rakov *et al.*'s work clearly elucidates the nanoscale solvation structure at charged alkali metal anodes in HCILE systems, shedding light on the mechanism by which these systems improve the cycling stability of alkali metal anodes.

Due to the positive charge dispersion around the imidazolium ring, which reduces electrostatic interactions between ion pairs, ILs based on $Emim^+$ cations typically exhibit lower viscosity. Sun *et al.* developed an EM-5Li-Na ILE (5 M LiFSI/0.16 M NaTFSI/EmimFSI) with high ionic conductivity of 2.6 mS cm^{-1}.[37] The prepared Li/LCO cell exhibited great electrochemical reversibility and long-cycle stability, because the formation of an F-rich SEI effectively suppresses the reduction of $Emim^+$ cations. Additionally, Na^+ ions may provide positively charged electrostatic shielding around the initial Li dendrite growth. Li/LCO cell retained about 81% of initial capacity after 1200 cycles at a high rate of 0.7C, with an average coulombic efficiency (CE) of approximately 99.9% (Fig. 4a). Liang *et al.* combined the IL 4.5 FSI-TFSI ILE (4.5 M LiFSI in $Py_{13}FSI$ + 1 wt% LiTFSI) with the low loading anode material (Cu@Si-PAN), and the Cu@Si-PAN/LNMO battery exhibited a specific capacity of 120 mA h g^{-1} and retained 80% capacity after 120 cycles (Fig. 4b).[38] Liu *et al.* synthesized $PMpyr_fFSI$ (1-methyl-1-propyl-3-fluoropyrrolidinium) *via* a one-step quaternization method, which can enter the inner-Helmholtz layer of electrodes with high Fermi levels, forming a stable EEI.[39] As shown in Fig. 4c, compared to non-fluorinated $PMpyr^+$, $PMpyr_f^+$ lowered the energy levels of the HOMO (Highest Occupied Molecular Orbital) and LUMO (Lowest Unoccupied Molecular Orbital), indicating that fluorinated cations are more resistant to oxidation and more readily reduced. The density functional theory (DFT) calculations showed that the C–F bond in $PMpyr_f^+$ is more prone to reductive decomposition than the C–H bond in $PMpyr^+$, resulting in dendrite-free Li deposition. Additionally, on the NCM622 ($LiNi_{0.6}Co_{0.2}Mn_{0.2}O_2$) cathode side, the deprotonation and ring-opening potential of $PMpyr_f^+$ was −0.49 eV, compared to −0.03 eV for $PMpyr^+$, indicating that the fluorinated cation is more oxidation-resistant, forming a stable and thin CEI that effectively suppresses electrolyte/cathode interfacial oxidation reactions (Fig. 4d).

2.2 Organic solvent-assisted ILE

Neat ILEs in LMBs possess some inherent advantages compared with traditional electrolytes, but several challenges remain. (i) Neat ILEs exhibit relatively low ionic conductivity at room temperature, limiting their applicability across a wide temperature range to a certain extent. (ii) Li^+ ions in neat ILEs with high viscosity show slow migration and poor rate performance. (iii) The reduction of IL organic cations can still form a potentially unstable EEI. (iv) The high preparation cost of ILs poses a challenge for large-scale commercial applications. Mixing ILs with organic solvents is a promising solution. For example, adding ester and ether solvents to ILEs can improve ionic conductivity, reduce viscosity and cost, while retaining the advantages of ionic liquids (Fig. 5a).

In our previous work, we explored the applicability of organic solvent-assisted ILEs in high-voltage LMBs at room temperature using *N*-ethyl-*N*-methyl-pyrrolidinium bis(fluorosulfonyl)imide

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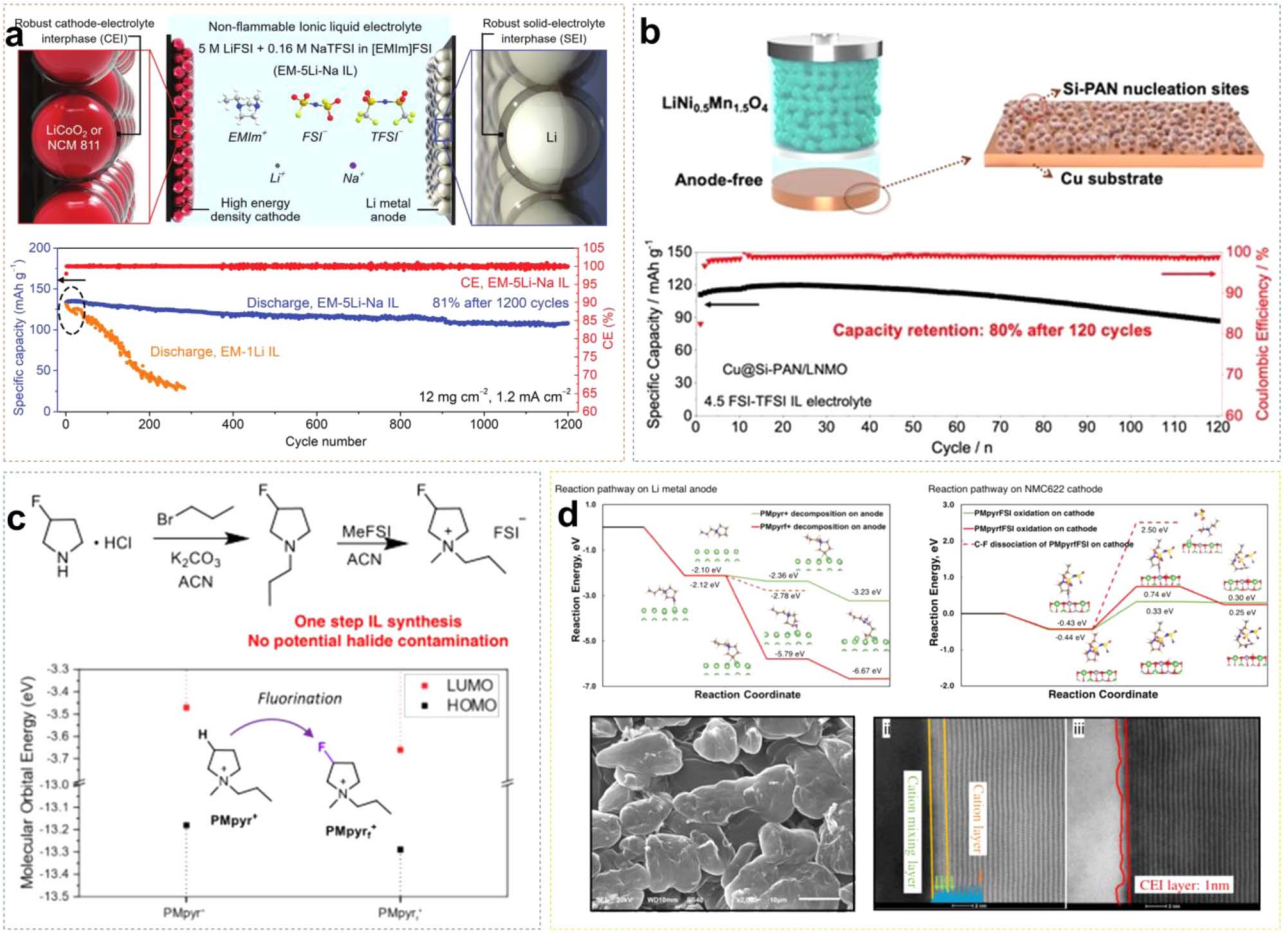

Fig. 4 (a) Schematic diagram of robust EEI formed by EM-5Li-Na ILE and corresponding cycling performance in Li/LCO cells with the mass loading of 12 mg cm^{-2}. Reprinted from ref. 37 with permission. Copyright 2020, Wiley. (b) Schematic illustrations of the high-voltage LNMO anode-free batteries using the Cu@Si-PAN current collector, and corresponding cycling performance of the LNMO anode-free batteries with 4.5 FSI-TFSI ILE. Reprinted from ref. 38 with permission. Copyright 2022, Wiley. (c) Synthesis route for PMpyr$_f$FSI and comparison of HOMO/LUMO energy levels for PMpyr$_f^+$ and PMpyr$^+$. (d) PMpyr$_f^+$ and PMpyr$^+$ reduction and oxidation pathway on Li metal and NMC622 cathode, respectively; the morphology of plated Li on Cu foil and HAADF-STEM analysis of cycled NMC622 cathodes. Reprinted from ref. 39 with permission. Copyright 2023, Springer Nature.

([C_2mpyr][FSI]) IL as the solvent and dimethyl carbonate (DMC) as the co-solvent.[40] In a low-concentration dual-anion electrolyte (LCDE) and moderate-concentration dual-anion electrolyte (MCDE), free DMC molecules and FSI$^-$ anions coordinate with Al^{3+}, dissolving the Al_2O_3 layer and continuously corroding the Al metal (Fig. 5b). However, in highly concentrated dual-anion electrolyte (HCDE), DMC and FSI$^-$ anions coordinate with Li^+, which can prevent the corrosion of the Al current collector. Compared to LCDE, HCDE exhibits excellent non-flammability. In addition, with the increase of LiTFSI salt concentration, free DMC solvent molecules coordinate with Li^+ ions. This coordination increases the energy required to break the solvation structure, significantly reducing the volatility of the electrolyte and enhancing thermal stability. DFT calculations reveal that in the HCDE system, LUMO energy levels are as follows: TFSI$^-$ < FSI$^-$ < DMC < C_2mpyr$^+$. Consequently, the SEI is primarily formed from the reductive decomposition of TFSI$^-$ and FSI$^-$ anions, and the main products of Li_3N, LiF, and Li_2O. A 4.5 V Li/NCM523 ($LiNi_{0.5}Co_{0.2}Mn_{0.3}O_2$) cell retains 95% of the initial capacity after 100 cycles (Fig. 5c).

Li *et al.* presented the synthesis of DFPyrTFSI (3,3-difluoro-*N*-methyl-*N*-propylpyrrolidinium bis(fluorosulfonyl)imide) IL, by using *gem*-difluorinated pyrrolidinium as a precursor and through a series of reactions, including the Eschweiler–Clarke reaction, quaternization, and anion-exchange (Fig. 5d).[41] DMC was mixed with DFPyrTFSI (mass ratio of 1 : 1) as a co-solvent, LiTFSI and LiDFOB (8 : 1 by mole) were added afterward to form a DMC-assisted ILE (2 M LiTFSI/LiDFOB in IL/DMC). Similar to the work by Liu *et al.*, the difluorinated IL cation exhibited lower HOMO and LUMO energy levels compared with non-fluorinated cations. More interestingly, DFT calculations suggest that the DFPyr$^+$ cation would be more prone to defluorination reduction at the anode, and could lead to the formation of a fluorine-rich SEI. A Li/NCM811 ($LiNi_{0.8}Co_{0.1}Mn_{0.1}O_2$) cell retained 90.5% of the initial capacity after 200 cycles, and an ultrahigh Ni-rich Li/NCM88 cell demonstrated excellent cycling stability, retaining 90.3% of the initial capacity

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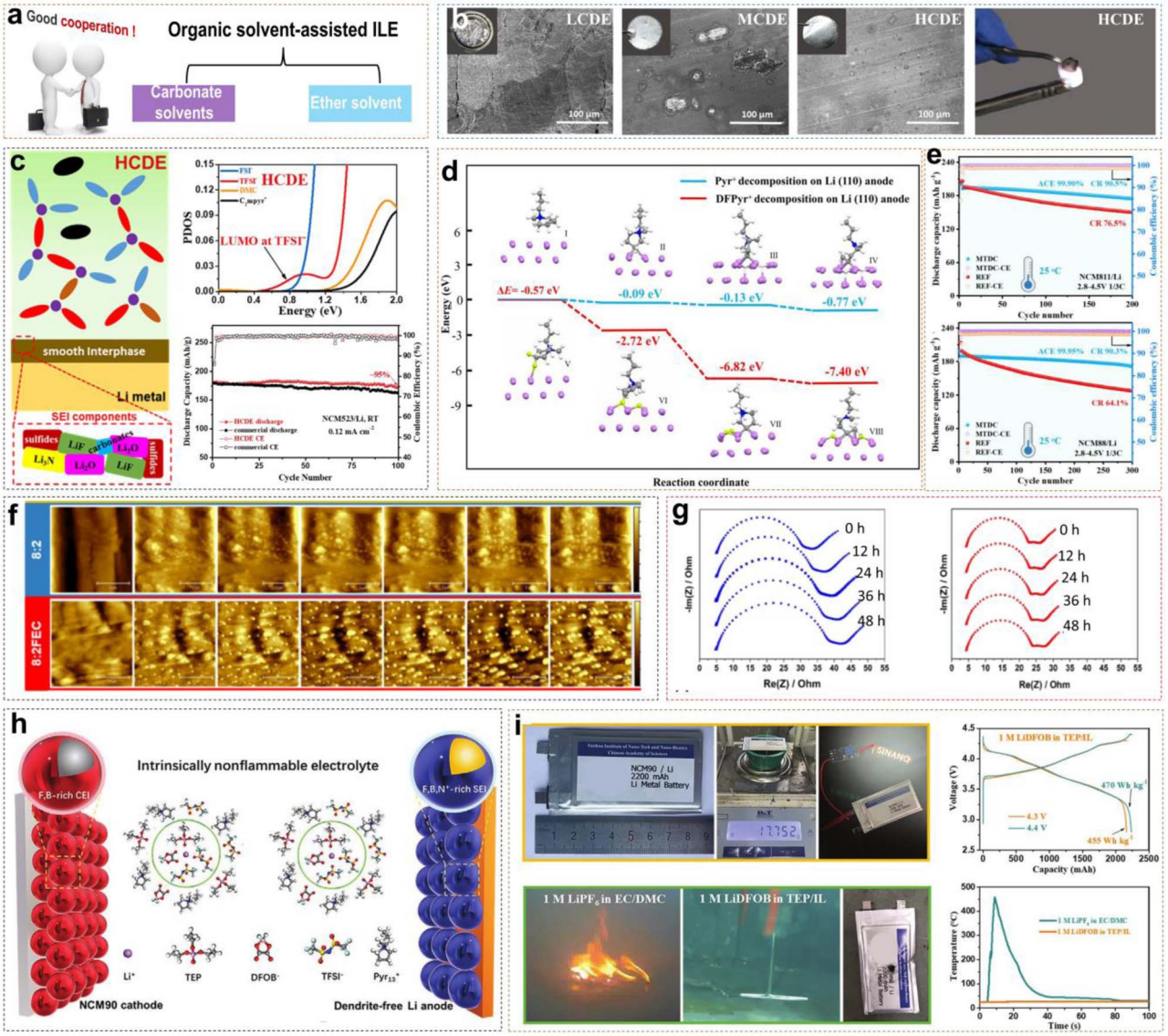

Fig. 5 (a) Classification of organic solvents for organic solvent-assisted ILEs. (b) Corrosion of Al foil in different electrolytes and combustion tests of HCDE. (c) Schematic of the SEI formed on the LMA surface and PDOS obtained by DFT calculations with HCDE; corresponding cycling performance of Li/NCM523 LMBs. Reprinted from ref. 40 with permission. Copyright 2020, Elsevier. (d) Reduction pathway of difluorinated IL on Li metal. (e) Cycling stability and coulombic efficiency of Li/NMC811 and Li/NMC88 cells under 2.8–4.5 V cutoff voltage at 25 °C. Reprinted from ref. 41 with permission. Copyright 2023, Wiley. (f) *In situ* AFM mappings (area 5 × 5 μm^2) of lithium plated onto nickel current collectors at 80 $\mu A\ cm^{-2}$ employing 8 : 2, and 8 : 2FEC as the electrolytes. (g) EIS performed on Li_{Ni}/Li cells using 8 : 2 and 8 : 2FEC during a two-day rest period with spectra recorded every 12 h. Reprinted from ref. 42 with permission. Copyright 2023, American Chemical Society. (h) Schematic diagram of the Li/NCM90 LMBs and solvation structure with the intrinsically non-flammable electrolyte. (i) Nail penetration test, charge/discharge curves and temperature *versus* time of the Li/NCM90 pouch cells. Reprinted from ref. 43 with permission. Copyright 2023, Wiley.

after 300 cycles with a high average efficiency of 99.95%. Recently, Li *et al.* designed 4,4-difluoro-*N*-methyl-*N*-propylpiperidinium bis(fluorosulfonyl)imi-de ($DFP_{13}TFSI$) by adopting a similar strategy, and the result shows that the 40 wt% LiTFSI/1 wt% LiDFOB/$DFP_{13}TFSI$/DMC ILE can be used to assemble the Li/NCM88 cell to deliver a high-capacity retention of 91.4% after 300 cycles.[44]

Meanwhile, the nucleation efficiency of Li^+ ions is usually accelerated after incorporating co-solvents into ILEs. Stępień *et al.* observed the actual Li^+ nuclei growth on a nickel (Ni) current collector by *in situ* AFM mapping.[42] They selected target electrolytes consisting of Pyr_{14}FSI-LiTFSI with ratios of 8 : 2 and compared them with 5 wt% FEC added as a control (8 : 2FEC). As shown in Fig. 5f, both systems exhibited granular Li deposition. In the 8 : 2 system, the Li particles varied in size, leading to a rough surface with the continuous growth of new Li nuclei. Conversely, in the 8 : 2FEC system, the Li particles were uniformly distributed and consistent in size, with no new nuclei appearing after 10 minutes. The above results suggested that FEC restrained the reductive decomposition of the electrolyte

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and induced smoother and denser Li deposition, which could reduce and stabilize the lithium nucleation overpotential significantly. When Li was deposited on a Ni current collector to form a Li_{Ni}/Li battery, impedance measurements taken every 12 h showed that the cell impedance increased over time in the 8 : 2 system. In contrast, the system with FEC remained relatively stable, indicating a more stable interface after initial Li plating in the 8 : 2FEC system (Fig. 5g). Unfortunately, the presence of flammable organic solvents like DMC and FEC reduces the safety of ILEs. Designing organic solvent-assisted ILEs that are both high-voltage and safe presents a significant challenge. Recently, we developed a triethyl phosphate (TEP)-assisted ILE using 1 M lithium difluoro(oxalato)borate (LiDFOB) in TEP and [Pyr_{13}][TFSI] IL.[43] Benefitting from high ionic conductivity, good thermal stability, and excellent passivation ability of LiDFOB on Al current collectors, we chose it as the Li salt. Particularly, TEP solvent is non-flammable, low viscosity, low cost, but reacts severely with metallic Li during charge–discharge cycles. Non-flammable ILs bring a large amount of Pyr_{13}^+ cations and $TFSI^-$ anions into the electrolyte. The Pyr_{13}^+ cations can ensure a uniform Li^+ ion flux distribution on the electrode surface by electrostatic shielding. Meanwhile, $TFSI^-$ and $DFOB^-$ anions would participate in the Li^+ solvation structure to form F- and B-rich CEI/SEI layers on the high-nickel cathode or LMA by oxidizing or reducing preferentially. All the above suppress side reactions between the electrode and electrolyte effectively (Fig. 5h). A 2.2 Ah multi-layer Li/NCM90 pouch cell with non-flammable electrolytes has been achieved with high energy densities of 455 W h kg^{-1} at 4.3 V and 470 W h kg^{-1} at 4.4 V, besides passing the nail penetration test stably.

Ether-based electrolytes have greater resistance to reduction than ester-based electrolytes, so they have been able to form thinner SEI layers on the anode and possess higher initial coulombic efficiency. Using ether-based electrolytes as co-solvents with ILs improves Li^+ ion transport kinetics in ILEs while overcoming their inherent oxidative instability. Pal *et al.* reported an ether-aided 80IL20DME electrolyte (3.2 mol kg^{-1} LiFSI in C_3mpyrFSI : DME = 80 : 20 wt : wt), demonstrating excellent anode stability and high voltage tolerance.[45] Fig. 6a shows SEM (scanning electron microscopy) images of Li morphology after cycling in both 80IL20DME and 100IL (3.2 mol kg^{-1} LiFSI in C3mpyrFSI) systems. After 10 and 100 cycles, obviously larger and denser Li metal particles were deposited in the DME-assisted ILE, and verified by Li deposited on a Cu substrate simultaneously. However, the 100IL and 80IL20DME systems also possessed rod-like deposits and block-

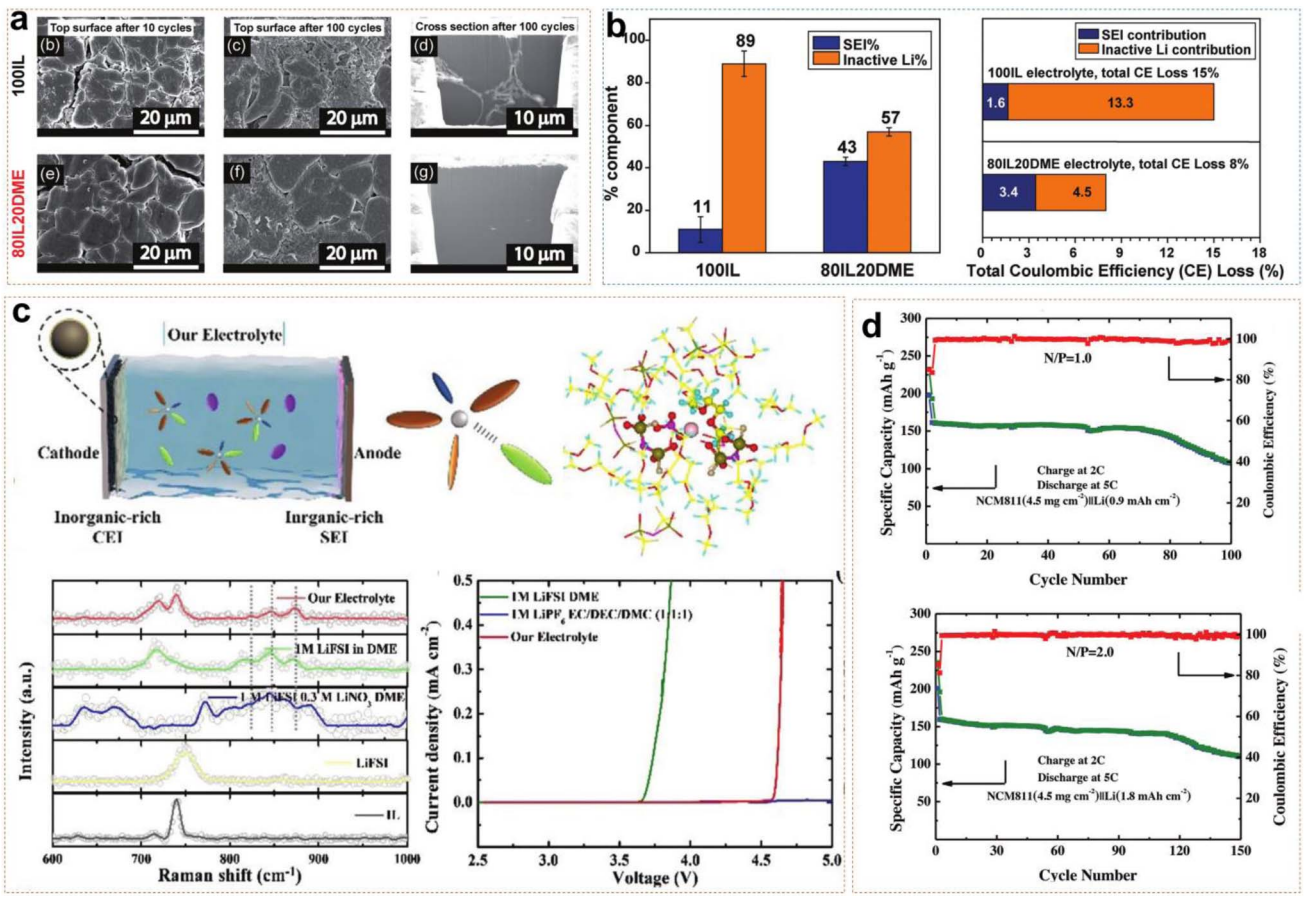

Fig. 6 (a) Morphology of the top surface and cross-section using Cryo-FIB in 100IL and 80IL20DME electrolyte. (b) Quantitative analysis of $Li_{inactive}^0$ and Li_{SEI}^+ by the TGC quantification method. Reprinted from ref. 45 with permission. Copyright 2022, The Royal Society of Chemistry. (c) Schematic illustration of Li/NCM811 full cells in IL-based electrolytes; Raman spectra, and linear sweep voltammetry curves. (d) Cycling performance of the Li/NCM811 full cell under N/P ratios of 2 and 1. Reprinted from ref. 46 with permission. Copyright 2023, Wiley.

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like deposits, respectively. Titration gas chromatography (TGC) quantification revealed that the SEI formation resulted in an 11% Li loss, with approximately 89% of the loss attributed to dead Li formation in the 100IL system. Meanwhile, the active Li loss was 43% and 57% in the 80IL20DME system, which were attributed to the SEI and dead Li, respectively (Fig. 6b). The results indicate that DME-assisted ILEs form a greater uniform and denser protective SEI than 100IL. In Li/Li symmetric cells, 80IL20DME exhibited lower polarization voltage compared to the 100IL. The Li/80IL20DME/NMC811 cell maintained an average coulombic efficiency of about 99.8% and a capacity retention of 81% after 300 cycles. Additionally, the Li|80IL20DME|NMC622 cell maintained an average coulombic efficiency of 99.23% after 100 cycles, based on the active material loading, which was as high as 22.4 mg cm^{-2} in the cathode. All the above results demonstrated the efficacy of ether-aided ILs in forming stable SEI layers and improving overall battery performance. Ding *et al.* dissolved 1 M LiFSI and 0.3 M $LiNO_3$ in a solution of $PP_{13}TFSI$ and DME with a volume ratio of 1 : 4.[46] Raman spectroscopy revealed that FSI^-, $TFSI^-$, and NO_3^- anions formed anion-rich solvation structures with Li^+ ions and solvent molecules (Fig. 6c). Meanwhile, the reduction of free solvent molecules enhanced the oxidative stability of the electrolyte system and promoted the formation of an inorganic-rich EEI. Linear sweep voltammetry (LSV) results indicated that the electrochemical window of the ether-based ILE was extended to 4.5 V. Additionally, DME, $LiNO_3$, and IL have higher HOMO energy levels which make it more likely to decompose on the NCM811 surface and form a CEI layer. The 4.3 V Li/NCM811 cells still exhibited stable cycling for 150 and 100 cycles, even though they had low N/P ratios of 2 and 1 (Fig. 6d). This demonstrates that the combination of LiFSI, $LiNO_3$, $PP_{13}TFSI$, and DME in the electrolyte formulation effectively enhances the stability and performance of LMBs even under challenging conditions.

2.3 Locally concentrated ILE

Although organic solvents (such as ethers and esters) can reduce the viscosity of ILs and improve ionic conductivity in batteries, they also introduce new challenges, such as poor thermal stability and safety, electrochemical instability, and strong side reactions. Ether and ester solvents usually evaporate or decompose under high-temperature conditions due to their low boiling points and high volatility, which increases the risk of thermal runaway. Additionally, organic solvents may lead to the formation of uneven and unstable SEI layers on the LMA surface. Thus, organic solvent-assisted ILE is in contrast to our original design concept of safety. In particular, for ester solvents, a thick and poorly conductive SEI formed from the decomposition products, namely, alkyl carbonates, which increase the interfacial impedance.[47] Furthermore, ether solvents are prone to decomposition under high voltage conditions, which makes them incompatible with high-voltage cathode materials.[48] The concept of locally concentrated ILE (LCILE) is similar to the classical LHCE (localized high concentration electrolyte), but both of them still have some subtle differences.[49] For the LHCE, only a salt concentration above a threshold (typically >3 mol kg^{-1}), can reduce the free solvent content and increase the content of contact ion pairs (CIPs) and aggregates (AGGs) significantly.[50] In contrast, ILs contain abundant film-forming anions, so the Li salt concentration in concentrated ionic liquid electrolytes (CILEs) is usually less than 2 mol kg^{-1}.

The basic design principles of LCILE include the following aspects: (i) the CILEs are formed by dissolving 1.5–2 mol kg^{-1} of Li salts into neat ILs. This leads to the creation of abundant anionic aggregate structures, which significantly enhance the stability of the EEI. (ii) The selection of non-polar or weakly polar co-solvents ensures that the original solvation structure of the CILEs is not disrupted. Additionally, the co-solvents should possess a wide electrochemical window and high thermal stability to maintain the LCILE's stability under high-voltage and high-temperature conditions. (iii) Adjusting the component ratios and Li salt concentrations helps maintain appropriate electrolyte viscosity while improving ionic conductivity, ensuring that the Li^+ ion kinetics in the LCILE system meet the requirements for its applications.

Fig. 7a shows the advantages and disadvantages of the traditional ether and carbonate cosolvents in CILEs. We can discover that LCILEs demonstrate significant advantages in terms of redox stability, viscosity, flammability, liquid range, and film-forming quality by comparison. Additionally, the composition of LCILEs is simple, usually consisting of a Li salt, IL, and cosolvent. Fig. 7b summarizes some recent reports about the compositions of LCILEs. LCILEs use LiFSI, LiTFSI, and LiDFOB as Li salts, sulfonamide-based ILs as solvents, and fluorinated ethers, aromatic molecules, and chlorinated molecules as cosolvents. In this section, we will explore the solvation structure and interfacial chemistry of LCILEs, and summarize their latest advancements in high-energy-density LMBs.

Fluorinated ether cosolvents not only enhance Li^+ ion conductivity and promote rapid charge–discharge performance by reducing the viscosity of the electrolyte, but also ensure a wide electrochemical window to make LCILEs operate at high voltages stably. In our previous work, we proposed an LCILE based on non-flammable $PP_{13}FSI$ and hydrofluoroether diluents (HFEs).[51] HFEs address the inherent drawbacks of ILs, such as high viscosity, poor separate wettability, low ionic conductivity, and high cost, while maintaining the flame-retardant properties of the electrolyte and enhancing battery safety. In ILE systems, the strong coulombic interactions between cations and anions result in high viscosity and poor Li^+ ion transport kinetics, leading to uneven Li deposition on the LMA surface. The addition of HFE accelerates Li^+ ion transport while preserving the anion-rich solvation structure, ensuring the formation of a stable inorganic SEI layer (Fig. 8a). As a result, the assembled Li/Cu cell can cycle stably for 800 cycles with a coulombic efficiency of 99.4%. The Li/LFP cell can cycle for 1000 cycles at a high rate of 5C, with a capacity retention rate of 87%. Compared to single anions, the synergistic effect of FSI^- and $TFSI^-$ dual anions further enhances SEI stability. Lee *et al.* designed an LCILE with LiTFSI as the lithium salt, $Pyr_{13}FSI$ IL as the solvent, and TTE as the cosolvent (1 : 2 : 2 by mol).[56] The Li/

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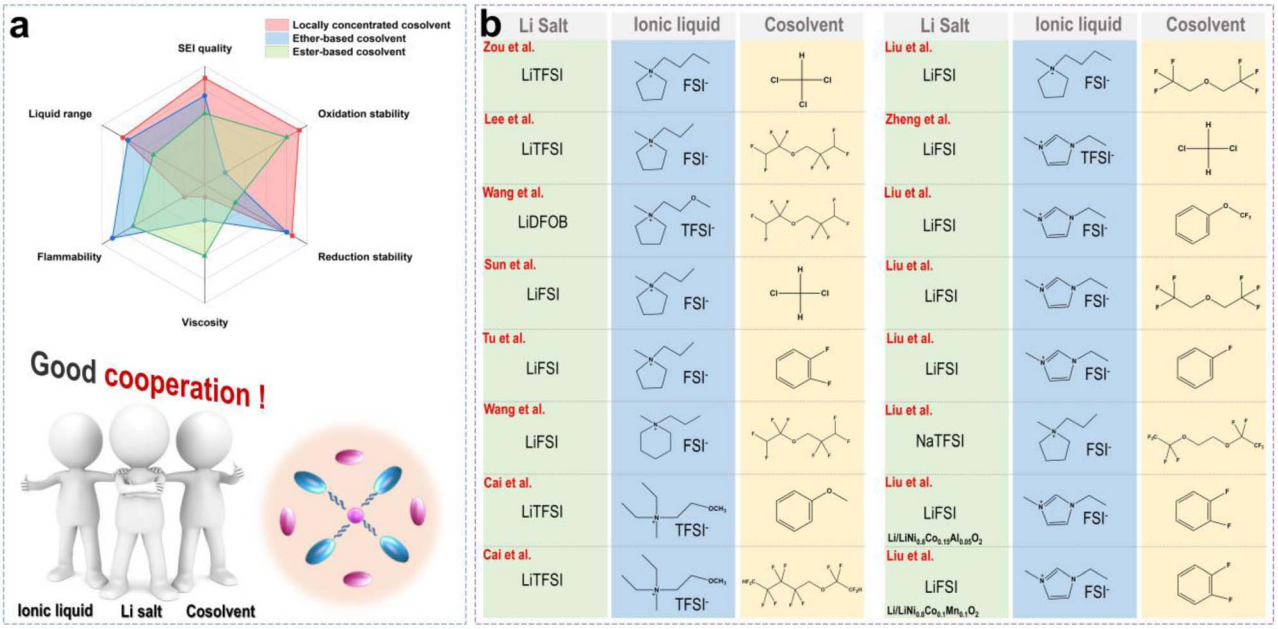

Fig. 7 (a) Comparison of the physicochemical and electrochemical properties of ILEs using an ether-based cosolvent, an ester-based cosolvent, and a locally concentrated cosolvent in a radar chart; and classical LCILE solvation structure. (b) Summary of the combination and collocation of LCILE composed of a lithium salt, cosolvent and ILs.

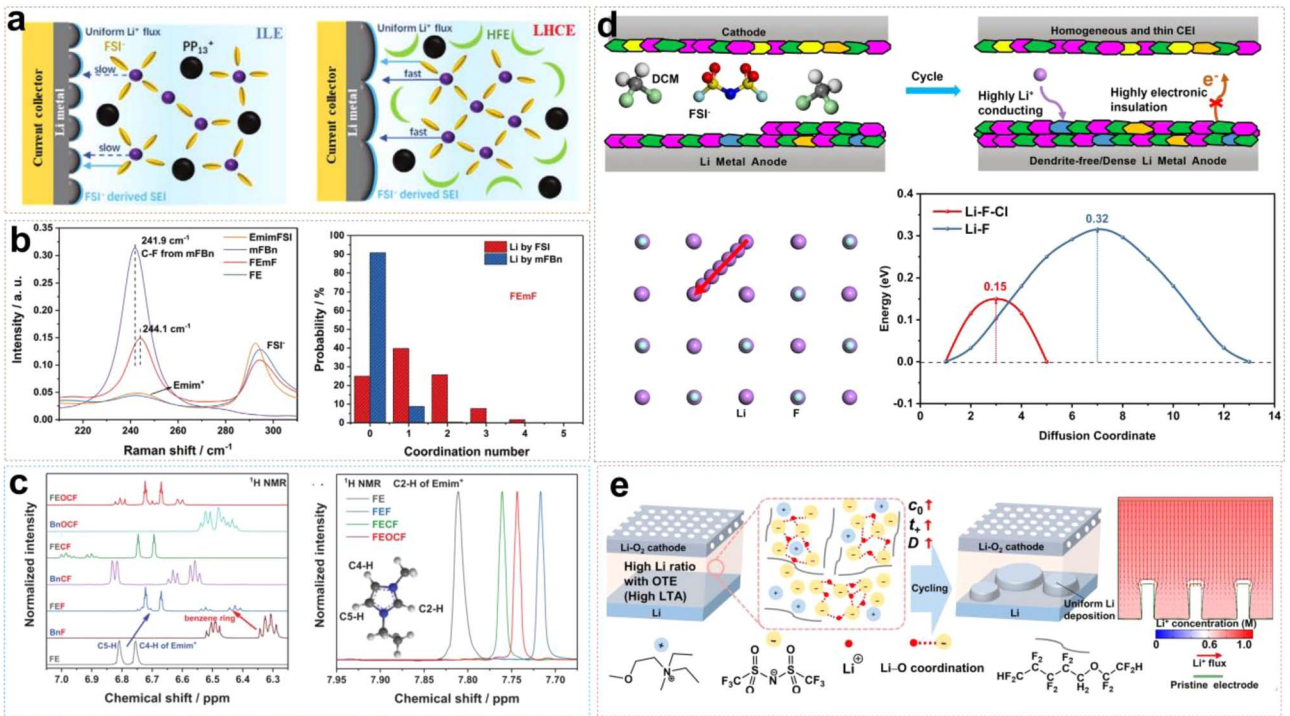

Fig. 8 (a) Schematic diagrams of Li plating on a Cu current collector and solvation structures of ILE and LHCE. Reprinted from ref. 51 with permission. Copyright 2021, Wiley. (b) Li^+ ion solvation structure with mFBn and the corresponding Raman characterization and coordination number calculation. Reprinted from ref. 52 with permission. Copyright 2022, Wiley. (c) 1D ^1H NMR spectra of the electrolytes and their constituents in the region of 6.25–7.05 ppm and 7.675–7.950 ppm. Reprinted from ref. 53 with permission. Copyright 2023, Wiley. (d) Schematic illustrations of the SEI/CEI components from LCM systems and Li^+ diffusion path and migration energy barriers of Li^+ along LiF–LiCl grain boundaries. Reprinted from ref. 54 with permission. Copyright 2024, American Chemical Society. (e) Schematic diagram of Li deposition behavior by regulating T_{sand} with OTE. Reprinted from ref. 55 with permission. Copyright 2021, Wiley.

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LCO cell shows outstanding cycling performance with around 80% of capacity retention after 400 cycles. Meanwhile, we also found that LCILE based on LiDFOB and *N*-methyl-*N*-methoxyethyl-pyrrolidinium bis(trifluoromethylsulfonyl)imide (MEMPTFSI) IL could maintain the lithium salt concentration at an extremely low level of 0.1 mol L^{-1}. This design of ultralow-concentration electrolyte (ULCE) employs LiDFOB as the lithium salt, which has excellent corrosion inhibition properties for aluminum foil and preferentially decomposes to form a stable, low-impedance SEI layer.[57] The designed SEI effectively suppresses lithium dendrite growth and improves the high and low-temperature performances of lithium batteries. At −40 °C, Li/NCM622 batteries with commercial electrolytes fail to operate properly, while batteries with ULCE retain a discharge capacity of 115 mA h g^{-1}. Even at −60 °C, the Li/NCM622 battery maintains 57% of room temperature capacity. Under high-temperature conditions of 70 °C, Li/NCM622 with ULCE can cycle for 150 cycles stably. In addition, organic cations play a significant role in LCILEs. Liu *et al.* discovered that $Emim^+$ cations coordinate with FSI^- less frequently compared to Pyr_{14}^+, resulting in lower viscosity and enhanced Li^+ ion transport. Furthermore, the SEI formed with $Emim^+$ is relatively more stable, due to its high nitrogen content.[58]

Fluorinated aromatic compounds are also excellent diluents for LCILEs, given their low affinity for Li^+ and superior F-donating ability compared to HFE. 1,2-Difluorobenzene (1,2-dfBen) was selected as a diluent for ILE due to its extremely low dielectric constant and abundance of fluorinated functional groups. We found that the inclusion of 1,2-dfBen increases the AGG content in the ILE system, attributed to the enhanced interaction between Li^+ and FSI^- in the low dielectric environment.[59] The abundance of AGG induces the formation of a robust SEI, significantly improving Li^+ ion transport at the interface and stabilizing the LMA. The 1,2-dfBen diluent is also suitable for high-voltage cathode materials. Liu *et al.* designed an LCILE composed of LiFSI, EmimFSI, and 1,2-dfBen in a molar ratio of 1 : 2 : 2, which achieved stable cycling of high-voltage $LiNi_{0.8}Co_{0.15}Al_{0.05}O_2$ and $LiNi_{0.8}Mn_{0.1}Co_{0.1}O_2$ cathodes with a mass loading of 10 mg cm^{-2}.[60,61] Unlike 1,2-dfBen, monofluorobenzene (mFBn) possesses a single fluorine functional group, which can solvate Li^+ ions partially. The self-diffusion coefficient of Li^+ in the LCILE with mFBn (FEmF) is 3.5×10^{-11} m^2 s^{-1}, 2.5 times higher than in pure ILE (FE).[52] As shown in Fig. 8b, the C–F stretching vibration of mFBn shifts from 241.9 cm^{-1} to 244.1 cm^{-1}, indicating the involvement of mFBn in the solvation structure. MD simulations of FEmE show an additional peak at 2.22 Å in the Li–F (mFBn) curve, suggesting that mFBn enters the solvation sheath of Li^+. Although the participation of mFBn in Li^+ solvation is limited, it reduces the Li^+–FSI^- interaction, lowering the viscosity and enhancing the ionic conductivity. Fluorinated aromatic compounds can also influence the π–π interactions between the benzene ring and $Emim^+$ cations and the coordination between Li^+ and cosolvent F atoms, thereby affecting the physicochemical properties of the electrolyte system. Liu *et al.* compared monofluorobenzene (BnF), trifluoromethylbenzene (BnCF), and trifluoromethoxybenzene (BnOCF) as diluents, finding that BnOCF promotes near-complete decomposition of FSI^- and $Emim^+$, enhancing the reversibility of LMA.[53] Raman spectroscopy analysis showed limited influence of BnOCF on Li^+ and FSI^- coordination. However, ^1H NMR spectroscopy revealed that the peaks of $Emim^+$ C5–H and C4–H shifted to lower frequencies, while the cosolvent peaks shifted to higher frequencies, indicating π–π interactions between $Emim^+$ and the fluorinated aromatic cosolvent (Fig. 8c). However, the environmental and occupational safety of perfluoroalkyl and polyfluoroalkyl substances (PFAS) are of concern. Recently, Liu *et al.* reported LCILEs using anisole as a cosolvent.[62] Anisole not only enhances ion transport by inducing nanophase-separated solvation structures but also regulates the deposition of organic cations and anions on LMA and the conversion of FSI^- to LiF in the SEI. The coulombic efficiency improved from 99.19% in ILE to 99.71% in LCILE by optimizing the anisole content. This electrolyte enabled stable cycling for 400 cycles in Li/$LiFePO_4$ cells and 350 cycles in Li/SPAN cells, with a capacity retention of 90% at an N/P ratio of 1.5.

Besides fluorinated diluents, chlorinated diluents have emerged as promising candidates for LCILEs and benefit from their lower cost and superior flame-retardant properties. We introduced dichloromethane (DCM) into the ILE system, formulating a 0.68 M LiFSI/C_3mpyrFSI/DCM electrolyte (LCM).[54] Our research demonstrated that chlorinated diluents effectively modulate the solvation structure of the ILE, reducing its viscosity and enhancing the bulk Li^+ ion transport kinetics. Furthermore, the coordination between chlorinated diluents and IL induces the formation of a uniform dual-halide EEI. This EEI mitigates severe side reactions between the electrode and electrolyte under extreme conditions (high voltage, high temperature and low temperature) effectively, enabling stable operation of LMBs across a broad temperature range (−20 to 60 °C). Fig. 8d shows that unique dual-halide EEI not only inhibits the growth of lithium dendrites at the anode but also forms a stable CEI. Simulation results of Li migration paths indicate that the LiF–LiCl interface has a lower diffusion energy barrier than a single LiF interface. Zou *et al.* employed trichloromethane (TCM, $CHCl_3$) as a diluent to develop an anion-strengthened solvation ILE (ASILEs) system, composed of 1 mol L^{-1} LiTFSI Pyr_{14}FSI/TCM.[63] TCM improved flame retardancy and enhanced the safety performance of LMBs. The assembled Li/NCM811 battery retained 81.6% of its initial capacity after 500 cycles at a 1C rate, and the Ah-level Li/NCM811 pouch cell achieved a high energy density of 386 W h kg^{-1}. The results demonstrate that the chlorinated diluents are feasible diluents in LCILE systems. LCILEs also have extensive applications not only in LMBs with insertion-type cathode materials but also Li–O_2 batteries. In Li–O_2 batteries, LCILEs stabilize O_2^- intermediates, enhancing reaction kinetics and reducing overcharge potentials in the O_2/O_2^- redox coupling. Although the use of *N*,*N*-diethyl-*N*-methyl-*N*-(2-methoxyethyl)ammonium bis(trifluoromethanesulfonyl)imide ([DEME][TFSI]) IL as a solvent in ILEs can lower the overpotential of Li–O_2 batteries to 0.68 V, the limiting current density at the Li anode remains low. As shown in Fig. 8e, Cai *et al.* assessed Li^+ transport capabilities using Sand's time

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Table 1 Summary of recent developments in IL-based electrolytes including neat ILE, organic solvent-assisted ILE, and LCILE

Electrolyte composition	Molar ratio or concentration	Battery type	Mass loading (mg cm^{-2}) and N/P ratio	Cycling stability (voltage, CR, cycles, rate, CE)	Ref.
LiTFSI : $Pyr_{14}FSI$	1 : 4	Li/NCM88	2.8, excessive Li	3.0 V–4.3 V, 88%, 1000 cycles, 0.3C, 99.94%, RT	31
LiTFSI/[C_6Py][TFSI]	1 mol L^{-1}	Li/LNMO	2.7, excessive Li	3.5 V–4.9 V, 70.2%, 55 cycles, 0.1C, 97%, 60 °C	33
LiFSI : [C_{1O1}mpip][FSI]	1 : 1	Li/LFP	3.82, excessive Li	2.5 V–3.8 V, 98.8%, 100 cycles, 0.2C, >99%, 50 °C	34
LiTFSI/NaTFSI/EMImFSI	5 mol L^{-1}	Li/LCO	10, 2	2.8 V–4.3 V, 90%, 140 cycles, 0.7C, 99.9%, RT	37
		Li/NCM811	10, 1.8	2.8 V–4.4 V, 95%, 120 cycles, 0.5C, 99.8%, RT	
LiFSI/$Py_{13}FSI$ + 1 wt% LiTFSI	4.7 mol L^{-1}	Cu@SiPAN/LMNO	10, anode-free	3.0 V–4.85 V, 80%, 120 cycles, 50 mA g^{-1}, 99%, RT	38
LiFSI : $PMpyr_1FSI$	1 : 1	Li/NCM622	8.8, 3	3.0 V–4.6 V, 98.5%, 100 cycles, 0.3C, 99%, RT	39
LiTFSI : [C_2mpyr][FSI] : DMC	13 : 4 : 13	Li/NCM523	4, excessive Li	2.8 V–4.5 V, 95%, 100 cycles, 0.12 mA cm^{-2}, 99%, RT	40
		Li/LFP	2, excessive Li	2.5 V–4.2 V, 80%, 500 cycles, 0.15 mA cm^{-2}, 99.9%, RT	
LiDFOB : TEP : [Pyr_{13}][TFSI]	10 : 29 : 17	Li/NCM90	15, 3	2.8 V–4.4 V, 99%, 50 cycles, 0.1C, 99%, RT	43
LiFSI/$LiNO_3$/PP_{13}TFSI/DME	1 mol L^{-1}	Li/NCM811	4.5, 1	3.0 V–4.3 V, 66%, 100 cycles, 2C, >99%, RT	46
LiFSI/C_3mpyrFSI/DME	3.2 mol kg^{-1}	Li/NCM811	6.4, 50 μm Li foil	2.8 V–4.4 V, 81%, 300 cycles, 0.2C, 99.8%, RT	45
LiTFSI/LiDFOB/DMC/DFPyrTFSI	8 : 1 : 20 : 4	Li/NCM811	5, excessive Li	2.8 V–4.5 V, 90.5%, 200 cycles, 0.3C, 99.9%, RT	41
		Li/NCM88	5, excessive Li	2.8 V–4.5 V, 90.3%, 300 cycles, 0.3C, 99.95%, RT	
LiTFSI/LiDFOB/DMC/DFPTFSI	40 : 3 : 60 : 12	Li/NCM811	5, excessive Li	2.8 V–4.5 V, 93.6%, 200 cycles, 0.2C, >99%, RT	44
		Li/NCM88	5, excessive Li	2.8 V–4.5 V, 91.4%, 300 cycles, 0.2C, >99%, RT	
LiFSI/$Pyr_{13}FSI$/1,2-dfBen	8 : 18 : 15	Li/LFP	9.5, excessive Li	2.5 V–4.0 V, 96%, 250 cycles, 0.5C, >99.6%, RT	59
LiTFSI/$PP_{13}FSI$/TTE	1 : 2 : 2	Li/LCO	8.0, excessive Li	3.0 V–4.3 V, 80%, 350 cycles, 0.5C, >99.6%, RT	56
LiFSI/$Pyr_{13}FSI$/DCM	8 : 18 : 23	Li/LCO	11.5, 5.28	2.8 V–4.4 V, 80.6%, 70 cycles, 0.3C, 99%, RT	54
LiTFSI/$Pyr_{14}FSI$/TCM	1 mol L^{-1}	Li/NCM811	5, excessive Li	3.0 V–4.3 V, 81.6%, 500 cycles, 1.0C, 99.9%, RT	63
LiFSI/EmimFSI/BnOCF	1 : 2 : 0.55	Li/NCA	21, 1.2	2.8 V–4.4 V, 71%, 150 cycles, 0.1C, >99%, RT	53
LiFSI/EmimFSI/mFBn	1 : 2 : 2	Li/SPAN	2.7, 1.8	1.0 V–3.0 V, 71%, 250 cycles, 0.3C, >99%, RT	52
LiFSI/EmimFSI/dFBn	1 : 2 : 2	Li/NCM811	10, 1	2.8 V–4.4 V, 62%, 100 cycles, 0.3C, >99%, RT	60
		Li/NCM811	10, anode-free	2.8 V–4.4 V, 76%, 250 cycles, 0.3C, >99.5%, RT	
LiFSI/EmimFSI/dFBn	1 : 2 : 2	Li/NCA	10, excessive Li	2.8 V–4.4 V, 92.8%, 100 cycles, 0.1C, >99.5%, −20 °C	61
		Li/NCA	10, 2.35	2.8 V–4.4 V, 70%, 100 cycles, 0.1C, >99%, −20 °C	
LiFSI/$PP_{13}FSI$/HFE	1 : 2 : 4	Li/LFP	5, excessive Li	2.5 V–4.2 V, 87%, 1000 cycles, 5C, >99.5%, RT	51
LiDFOB/MEMPTFSI/HFE	3 : 34 : 132	Li/NCM622	5, excessive Li	2.8 V–4.5 V, 96%, 100 cycles, 0.2C, 99.9%, RT	57
LiFSI/EmimFSI/DCM	5 : 11.73 : 117	Li/LFP	N/A, excessive Li	2.5 V–4.0 V, 100%, 150 cycles, 0.1C, >99%, −40 °C	65
LiFSI/$Pyr_{14}FSI$/BTFE	3 : 4 : 4	Li/LFP	6.3, excessive Li	2.2 V–4.0 V, 94.6%, 400 cycles, 1C, >99.97%, RT	66
		Li/NCM523	7.6, excessive Li	2.8 V–4.3 V, 93.9%, 150 cycles, 0.3C, 99.91%, RT	

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Table 1 (*Contd.*)

Electrolyte composition	Molar ratio or concentration	Battery type	Mass loading (mg cm^{-2}) and N/P ratio	Cycling stability (voltage, CR, cycles, rate, CE)	Ref.
LiFSI/EmimFSI/ Anisole	1 : 2 : 6	Li/LFP	10, 1.5	2.4 V–3.6 V, 94%, 400 cycles, 0.3C, >99%, RT	62
		Li/SPAN	2.8 2.5	1.0 V–3.0 V, 90%, 350 cycles, 0.3C, >99%, RT	
LiFSI/EmimFSI/ BTFE	1 : 2 : 2	Li/NCM811	10, excessive Li	2.8 V–4.4 V, 96%, 200 cycles, 0.3C, 99.91%, RT	58
LiFSI/PP_{13}FSI/HFE	1 : 2 : 4	Li/Gr	2, excessive Li	0.01 V–1 V, 85%, 300 cycles, 3C, >99%, RT	67
LiTFSI/DEMETFSI/ MOP	0.95 : 1 : 3	Li/O_2	N/A, excessive Li	4 V charging voltage, 120 cycles, 0.16 V overpotential	64
LiTFSI/DEMETFSI/ OTE	0.8 : 1 : 4	Li/O_2	N/A, excessive Li	4 V charging voltage, 75 cycles, 0.064 V overpotential	55
NaTFSI/Py_{13}FSI/ TFEE	1 : 3 : 1	Na/NFM	1.3, excessive Na	2.0 V–4.1 V, 92%, 150 cycles, 0.5C, >99%, RT	68
$NaPF_6$/C_4C_1imBF_4/ diglyme	1 mol L^{-1}	Na/NVP	2, excessive Na	2.5 V–4.0 V, 90.7%, 1000 cycles, 2C, 99.6%, RT	69

(T_{sand}), which measures the duration during which Li^+ ions are depleted near a flat electrode surface without replenishment. By incorporating 1*H*,1*H*,5*H*-octafluoropentyl 1,1,2,2-tetrafluoroethyl ether (OTE) and anisole (MOP) to create LCILEs, the system maintained a high initial Li^+ ion concentration (C_0) while simultaneously reducing viscosity and increasing the Li^+ diffusion coefficient (D), thereby significantly extending T_{sand}.[55,64] These studies provide important theoretical and practical foundations for the development of novel LCILEs. Table 1 summarizes the properties of IL-based electrolytes designed by using IL as a solvent in recent years.

3 IL-based solid electrolytes

Owing to their excellent ion conductivity across a broad temperature range, liquid electrolytes have been widely acknowledged for over two decades.[70,71] Nevertheless, the risk of leakage and flammability along with the conventional organic liquid electrolytes also needs to be considered, so there is still a gap in the large-scale use of LMBs.[72–75] Compared to liquid electrolytes, solid electrolytes exhibit higher energy density and greater safety, which could have tremendous potential in batteries and other electrochemical devices.[76,77] Despite their numerous advantages, the main challenge faced by advanced application is the low ionic conductivity and slow Li^+ transfer kinetics.[78] Obviously, the large interfacial resistance originating from the insufficient solid–solid contact at the EEI always hinders the development of solid-state LMBs.[70]

In addition to their prominent role in liquid–electrolyte battery systems, ILs have emerged to be pivotal for promoting the development of solid-state electrolytes. Their practical use can be broadly classified into three key areas: (i) PILs act directly as polymer matrices through polymerization initiated by an initiator; (ii) utilization of ILs as plasticizers and charge carriers to enhance ionic conductivity and other properties of polymer electrolytes; (iii) incorporation of ILs to improve the interfacial contact in composite solid-state polymer electrolytes (CSPEs). The subsequent sections will discuss these three applications and clarify how ILs facilitate the development and performance of solid-state electrolyte systems.

3.1 PIL-based solid electrolytes

The PILs are composed of distinct IL monomers that are covalently linked together through the identical repeating unit (Fig. 9a).[79–82] The uniqueness of PILs lies in the integration of their exceptional properties (ILs) with the macromolecular structure of polymers.[81,83–85] Consequently, PILs possess superior properties such as electrochemical stability, thermal stability, and excellent mechanical strength by regulating the backbone structure.[86] As shown in Fig. 9b, we compared the main properties of solid-state electrolytes using PIL-based SPEs, oxide-based inorganic solid-state electrolytes (ISEs), and sulfide-based ISEs. Ideally, PILs would form self-supporting membranes with high conductivity and electrochemical stability that are able to dissolve substantial amounts of lithium salts. However, the polymerization of PIL monomers markedly elevates their glass transition temperature (T_g), consequently reducing their ionic conductivity by even several orders of magnitude.[87,88] Researchers have primarily adopted two strategies to address this issue: (i) designing novel PILs with complex macromolecular structures; (ii) introduction of modifications such as copolymers or plasticizers. This section will discuss the direct polymerization of neat PIL monomers and other modification approaches.

3.1.1 Direct polymerization of neat PIL monomers. Fig. 9c shows a schematic illustration of the PIL-based solid-state electrolyte transport model. Similar to the PEO, the ion transport in PIL-based electrolytes is generally coupled with local relaxation and segmental motion, and the kinetics of ions in the amorphous phases will be improved.[24] As mentioned previously, the ionic conductivity of the PIL monomers significantly decreases after polymerization. To enhance ionic conductivity

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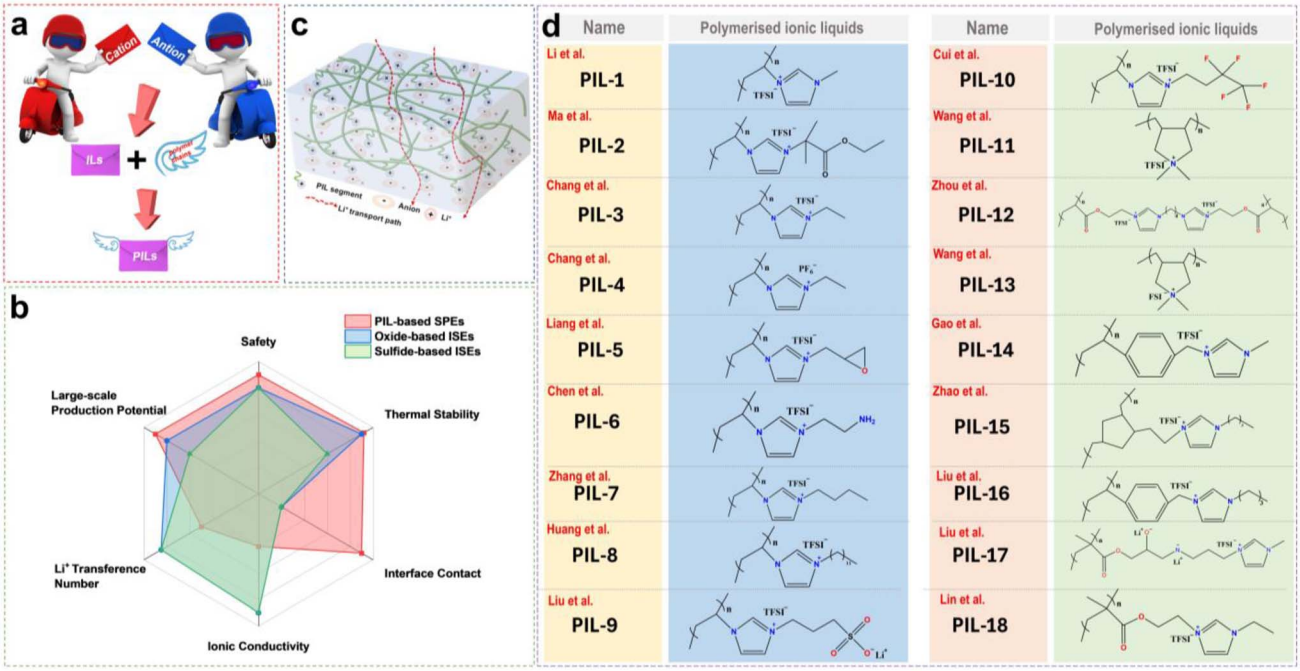

Fig. 9 (a) Schematic illustration of the relationship between ILs and PILs. (b) Comparison of the main properties of solid-state electrolytes using PIL-based SPEs, oxide-based inorganic solid-state electrolytes (ISEs), and sulfide-based ISEs in a radar chart. (c) Schematic illustration of PIL-based solid-state electrolyte transport model. (d) Summary of the chemical structures of the PILs used for solid-state electrolytes.

and elucidate the relationship between improved ion dynamics and alterations in ion interactions, Wang *et al.* have reported a class of solvent-free polymer electrolytes by using poly(diallyldimethylammonium bis(fluorosulfonyl)imide) ([PDADMA][FSI]) with a high content of lithium salt (LiFSI).[89] MD simulations demonstrate that the FSI^- anions coordinate with both the polycations and Li^+. This co-coordination reaches its peak at a molar ratio of 1 : 1.5 of [PDADMA][FSI] : LiFSI (Fig. 10a). Corresponding experimental data also indicated that increasing the lithium salt content facilitates Li^+ ion transport, as illustrated in Fig. 10b. When the salt content increases, more Li^+ ions start to co-coordinate to those FSI^- anions originally associated with the polycations only. This reduces the interactions between polycations and FSI^-anions, allowing greater freedom of movement for the polymer backbone, to raise the ionic conductivity. However, excessively high salt content can give rise to phase segregation and the formation of crystalline phases, consequently decreasing the number of "free" charge carriers and making low ion conductivity. The optimized PIL electrolyte is used in lithium symmetrical batteries and LMBs, exhibiting long-term stability and consistent cycling characteristics across varying current densities, thereby ensuring better performance (Fig. 10c).

Besides utilizing high-salt battery systems to achieve satisfactory performances, leveraging the structural adjustability of ILs and rationally designing the molecular structures of PIL monomers are other convenient and effective strategies.[90,92] This approach not only renders certain good electrochemical performances but also imparts unique properties to the polymers, such as hydrophobicity and self-healing capability. As shown in Fig. 10d, Lin *et al.* synthesized a novel PIL polymer electrolyte by grafting IL chain units ($EMIM^+$) into polymer backbones (poly(methyl methacrylate) (PMMA)).[90] PMMA, which is chemically stable with lithium metal, served as the matrix for the electrolyte, while the H^2, H^4, and H^5 protons in $EMIM^+$ acted as hydrogen bond donors, interacting with oxygen-containing groups in PMMA to form hydrogen bonds. This interaction enhanced the Li^+ transport and self-healing abilities of the polymer. Due to the external hydrogen bond interactions, the designed PIL could spontaneously heal cracks caused by dendrite growth at the EEI, resulting in uniform lithium deposition. As described in Fig. 10e, 91.2% of the LFP/PIL/Li battery capacity was retained after 206 cycles at RT (0.5C). When the temperature increased to 48 °C, 74.5% of the battery capacity was retained after 560 cycles, while CE stabilized near 100%. In summary, neat PILs have made significant strides as solid electrolytes. Nevertheless, the marked decrease in ionic conductivity following polymerization remains a pressing issue requiring resolution. Therefore, researchers must meticulously design PIL structures, incorporate additional functional groups for functionalization, and enhance their ionic conductivities while preserving their excellent mechanical properties.

3.1.2 Other modifications to PILs. The repetitive polar units in the backbones of PILs generate numerous crystalline regions contributing to a low ambient ionic conductivity.[92] Researchers have proposed several strategies for heightening the ambient ionic conductivity of PIL-based solid polymer electrolytes (SPEs), including copolymerization with various monomers,[84,91] introducing distinct polymers,[93] and introducing plasticizers.[92] As demonstrated in Fig. 10f, our research

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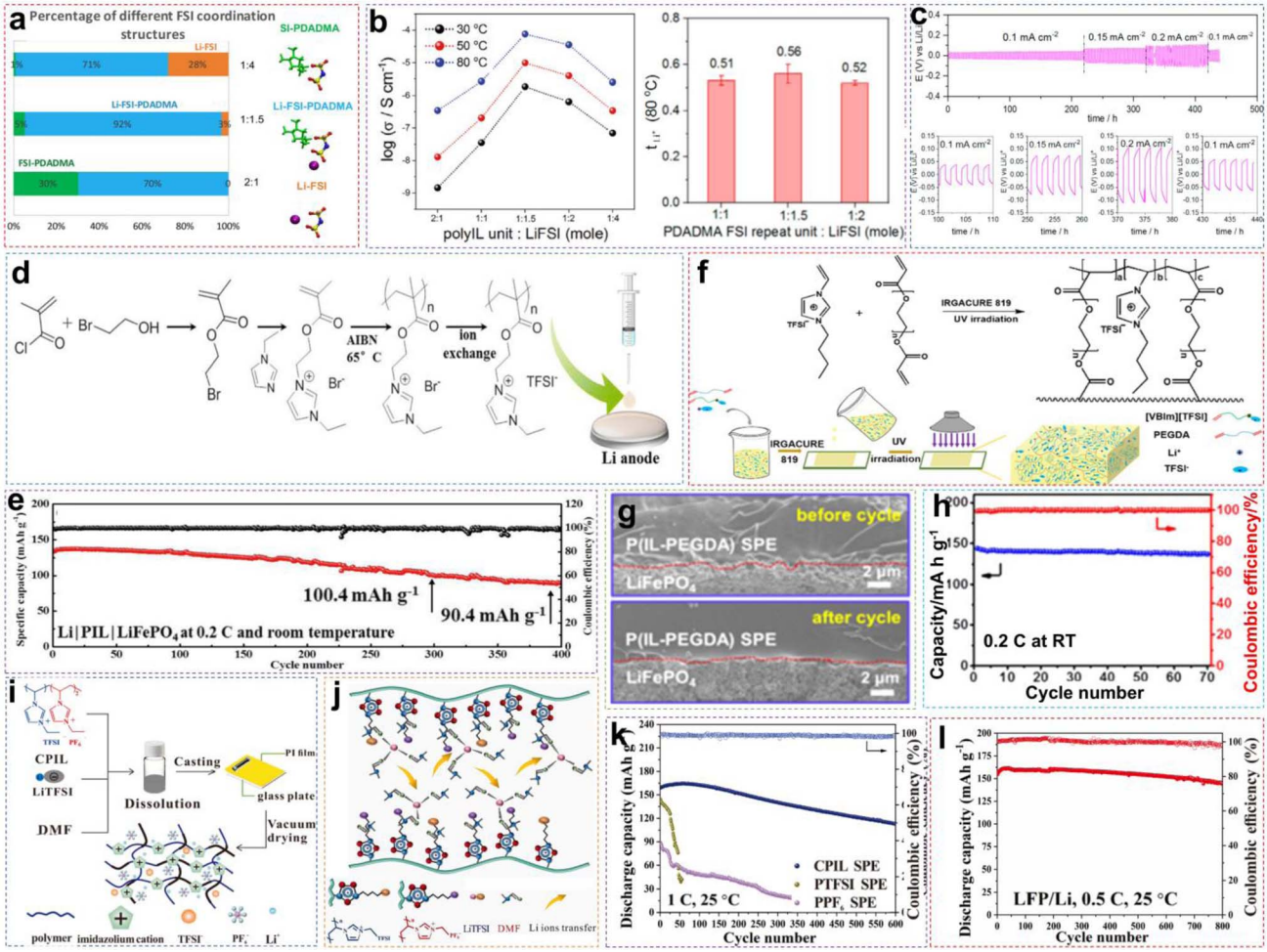

Fig. 10 (a) Percentage of anions in three types of FSI^- coordination environment calculated from MD simulations at 353 K. (b) The ion conductivities of different polyIL/salt systems at different temperatures and the Li^+ transference number of solid polymer electrolytes with varying polymer-to-salt ratios. (c) The Li/Li symmetric battery cycling performances of the 1 : 1 polymer electrolyte at various current densities (*i.e.*, 0.1, 0.15, and 0.2 mA cm^{-2}). Reproduced from ref. 89 with permission. Copyright 2019, Elsevier. (d) Schematic representation of the synthesis of PIL. (e) Cycling performance of LFP/PIL/Li batteries at 0.2C and RT. Reproduced from ref. 90 with permission. Copyright 2023, The Royal Society of Chemistry. (f) Schematic diagram of the P(IL-PEGDA) SPE copolymerization reaction and preparation process. (g) SEM image of the interface between the LFP cathode and P(IL-PEGDA) SPE before and after 50 cycles. (h) Cycling performance of LFP/P(IL-PEGDA) SPE/Li batteries at 0.2C and RT. Reproduced from ref. 91 with permission. Copyright 2020, American Chemical Society. (i) Schematic illustration of the fabrication of solid polymer electrolytes. (j) Schematic representation of Li^+ transfer in robust-flexible CPIL SPE. (k) Cycling performance of NCM811/CPIL SPE/Li batteries at 1C and RT. (l) Cycling performance of LFP/CPIL SPE/Li batteries at 0.5C and RT. Reproduced from ref. 84 with permission. Copyright 2024, Elsevier.

group synthesized an *in situ* plasticized IL-based SPE for LMBs by UV-curing 1-vinyl-3-butylimidazolium bis(trifluoromethylsulfonyl)imide ([VBIM]–[TFSI]) and poly(ethylene glycol) diacrylate (PEGDA) monomer.[91] The introduction of PEGDA *in situ* plasticized the PIL segments, reducing the crystallinity and glass-transition temperature (T_g) of the PIL, thereby facilitating polymer chain segmental movement and enhancing Li^+ ionic conductivity. Moreover, the electrostatic forces between the cationic IL moieties and the $TFSI^-$ anions, combined with the coordination between the oxygen atoms in PEGDA and Li^+, synergistically promoted the dissociation of lithium salts. As a result, the prepared P(IL-PEGDA) SPE exhibited high ambient temperature ionic conductivity (1.4×10^{-4} S cm^{-1}), high electrochemical window (5.0 V *vs.* Li^+/Li), non-flammability and thermal stability. Owing to the electrode/SPE interface (Fig. 10g), together with effective inhibition of dendrite growth, the solid-state LFP/P(IL-PEGDA) SPE/Li battery exhibited excellent cycling stability at 0.2C, with a reversible specific capacity of 140 mA h g^{-1} and CE of approximately 100% at RT (Fig. 10h).

Recently, Chang *et al.* produced a rigid–flexible PIL-based random copolymer by utilizing two PIL monomers with different anions.[84] This was accomplished by precisely adjusting the ratio of the soft segment (1-vinyl-3-ethylimidazolium bis(trifluoromethylsulfonyl)imide) ([VEIM][TFSI]) to the hard segment (1-vinyl-3-ethylimidazolium hexafluorophosphate ([VEIM][PF_6])) through making use of cross-linked copolymerization reactions (denoted as CPIL) (Fig. 10i). Notably, the large

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Table 2 Summary of recent developments in PIL-based electrolytes

Electrolyte composition	Conductivity (S cm^{-1})	ESW (V)	Battery type	Cycling stability (voltage, CR, cycles, rate, CE, temperature)	Ref.
LiTFSI/PEGDA/PEO/PIL-1 (poly[1-vinyl-3-methylimidazolium bis(trifluoromethylsulfo-nyl)imide])	4.3×10^{-5} (RT) 6.12×10^{-4} (55 °C)	5.44 V	Li/LFP	2.8 V–4.0 V, 77.6%, 50 cycles, 0.2C, >99%, 55 °C	94
LiTFSI/PEGMEM/PEGDA/EC/PC/PIL-2 (poly[1-vinyl-3-isobutyrate ethylimidazole bis(trifluoromethylsulfonyl)imide])	2.15×10^{-4} (30 °C)	~4.8 V	Li/LFP	2.7 V–4.0 V, 96%, 200 cycles, 0.1C, N/A, 60 °C	80
			Li/LCO	3 V–4.2 V, 92.1%, 100 cycles, 0.1C, N/A, RT	
LiTFSI/PIL-3 (poly[1-vinyl-3-ethylimidazolium bis(trifluoromethylsulfonyl)imide])/PIL-4 (poly[1-vinyl-3-ethylimidazolium hexafluorophosphate])	1.06×10^{-4} (RT)	4.5 V	Li/NCM811	2.7 V–4.3 V, 70.8%, 600 cycles, 1C, N/A, RT	84
			Li/LFP	2 V–4 V, 89.6%, 800 cycles, 0.5C, N/A, RT	
LiTFSI/EMIMTFSI/PEI/PVDF-HFP/PIL-5 (poly[1-vinyl-3-(oxiran-2-ylmethyl) imidazolium bis(trifluoromethylsulfonyl) imide])	1.8×10^{-3} (RT)	5 V	Li/LFP	2.5 V–4.2 V, 98%, 200 cycles, 0.5C, 98%, RT	86
			Li/NCM622	2.7 V–4.3 V, 98%, 50 cycles, 0.1C, N/A, RT	
LiTFSI/PEGDA/BA/EPPOSS/BMIMTFSI/PIL-6 (poly[amino-vinyl bifunctionalized imidazolium bis(trifluoromethylsulfonyl) imide])	2.5×10^{-4} (RT)	~5.1 V	Li/LFP	2.5 V–4.2 V, ~100%, 200 cycles, 0.2C, 98%, RT	95
LiTFSI/PEGDA/PIL-7 (poly[1-vinyl-3-butylimidazolium bis(trifluoromethylsulfonyl)imide])	1.4×10^{-4} (30 °C)	5 V	Li/LFP	2.5 V–4.2 V, 97%, 70 cycles, 0.2C, ~100%, RT	91
LiTFSI/PEGDMA/BMIMTFSI/VC/PIL-8 (poly[1-vinyl-3-dodecylimidazolium bis(trifluoromethanesulfonyl)imide])	7×10^{-4} (RT)	5 V	Li/LFP	2.8 V–4.2 V, 98.9%, 100 cycles, 0.1C, N/A, RT	82
PEGMEA/PEGDA/PIL-9 (poly[1-vinyl-3-(propylsulphopropyl)imidazolium bis(trifluoro-methanesulfonyl)imide])	1.1×10^{-5} (30 °C)	5.4 V	Li/LFP	2.5 V–4 V, 90.16%, 100 cycles, 0.2C, 99%, RT	96
LiTFSI/PVDF-HFP/GO-g-PIL-10 (poly[3-(3,3,4,4,4-pentafluorobutyl)-1-vinyl-1*H*-imidazole-3-ium bis(trifluoromethanesulfonyl)imide])	3.24×10^{-4} (RT)	~4.75 V	Li/LFP	N/A, 82%, 350 cycles, 0.5C, N/A, 30 °C	97
LiFSI/Pyr_{13} FSI/PIL-11 (poly[(diallyldimethylammonium) bis(trifluoromethanesulfonyl)imide])	1.7×10^{-5} (RT)	N/A	Li/NCM	3.0 V–4.3 V, 88.8%, 50 cycles, 0.05C, N/A, 50 °C	98
LiTFSI/EMITFSI/FEC/PIL-11/PIL-12 (poly-[1,4-bis[3-(2-acryloyloxyethyl)imidazolium-1-yl]butane bis[bis(trifluoromethanesulfonyl) imide]])	1.06×10^{-3} (RT)	~4.4 V	Li/LFP	2.4 V–4.2 V, 97.7%, 100 cycles, 0.1C, ~100%, RT	85
LiFSI/PIL-13 (poly[(diallyldimethylammonium) bis(fluorosulfonyl)imide])	7×10^{-5} (80 °C)	N/A	Li/LFP	2.5 V–3.8 V, N/A, 30 cycles, 0.067C, 99.94%, 80 °C	89
			Li/NCM	3.0 V–4.3 V, 67.5%, 50 cycles, 0.1C, 99.95%, 80 °C	
LiTFSI/EMIMTFSI/TEOS/PVDF-HFP/PIL-11	5.3×10^{-4} (20 °C)	4.9 V	Li/LFP	2.5–4.0 V, 95.7%, 250 cycles, 3C, 99.1%, 100 °C	99
LiTFSI/PEO/PVDF-HFP/PIL-14 (poly[1-(4-vinylbenzyl)-3-methylimidazolium bis(trifluoromethylsulfonyl)imide])	3.7×10^{-4} (RT)	5 V	Li/LFP	2.5 V–3.8 V, 97.1%, 250 cycles, 0.5C, N/A, RT	81
LiTFSI/PEG/PIL-15 (poly[(1-ethyl-3-(2-3-vinylcyclopentyl)ethyl)-1*H*-imidazol-3-ium bis(trifluoromethylsulfonyl)imide])	1.5×10^{-5} (30 °C)	4.6 V	Li/LFP	2.5 V–4.0 V, 92%, 70 cycles, 0.2C, N/A, 50 °C	100
LiTFSI/PEGDA/POSS/PIL-16 (poly[(1-(4-vinylbenzyl)-3-butylimidazolium bis(trifluoromethanesulfonyl)imide])	1.8×10^{-4} (30 °C)	~5 V	Li/LFP	2.5 V–4.0 V, 80%, 150 cycles, 0.5C, N/A, RT	101
LiTFSI/MEMPTFSI/PIL-17 (poly[dilithium mono(3-(methacryloyloxy)-2-oxidopropyl)-3-(1-methyl-1*H*-imidazol-3 ium-3-yl)prop-yl bis(trifluoromethylsulfonyl)imide])	4.3×10^{-4} (30 °C)	4.3 V	Li/LFP	2.5 V–4.0 V, 87.6%, 144 cycles, 0.2C, N/A, 30 °C	92
LiTFSI/PIL-18 (poly[(1-ethyl-3-(2-methacryloyloxy)ethyl)imidazolium bis(trifluoro methylsulphonyl)imide])	1.76×10^{-4} (25 °C)	5.2 V	Li/LFP	2.5 V–4.3 V, 91.2%, 206 cycles, 0.5C, N/A, RT	90

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volume and more delocalized structure of the $TFSI^-$ anion in the soft segment facilitate fast ion migration, while the hard segment provides excellent mechanical strength and excellent antioxidative properties. Furthermore, the residual DMF molecules in the form of $[Li(DMF)_x]^+$ acted as a polar solvent, softening the PTFSI and PPF_6 chains, which allowed the dissociated Li^+ to be transported among the CPIL (Fig. 10j). The combination of the above factors led to the establishment of an effective Li^+ transport pathway in SPE, which was made possible for the rational regulation of the ratio of the two components, so that an ionic conductivity of 1.06×10^{-4} S cm^{-1} and a stable electrochemical window up to 4.5 V (*vs.* Li^+/Li) at RT could be satisfied. Based on the above advantages, the solid-state NCM811/CPIL SPE/Li batteries put up a stable cycling behavior, maintaining a high CR of 70.8% after 600 cycles at 1C (Fig. 10k). Besides, the LFP/CPIL SPE/Li batteries demonstrated a discharge capacity of 144.7 mA h g^{-1}, with a CR of 89.6% after 800 cycles at 0.5C (Fig. 10l).

For polymer electrolytes comprising PILs as the matrix, the most prevalent approach to enhancing the ionic conductivity is to select ILs devoid of polymerization sites as plasticizers to optimize the overall performance of the PILs, which will be formulated in conjunction with this topic in the subsequent section. Furthermore, Fig. 9d summarizes the structural types of PILs employed in PIL-based solid-state electrolytes in recent

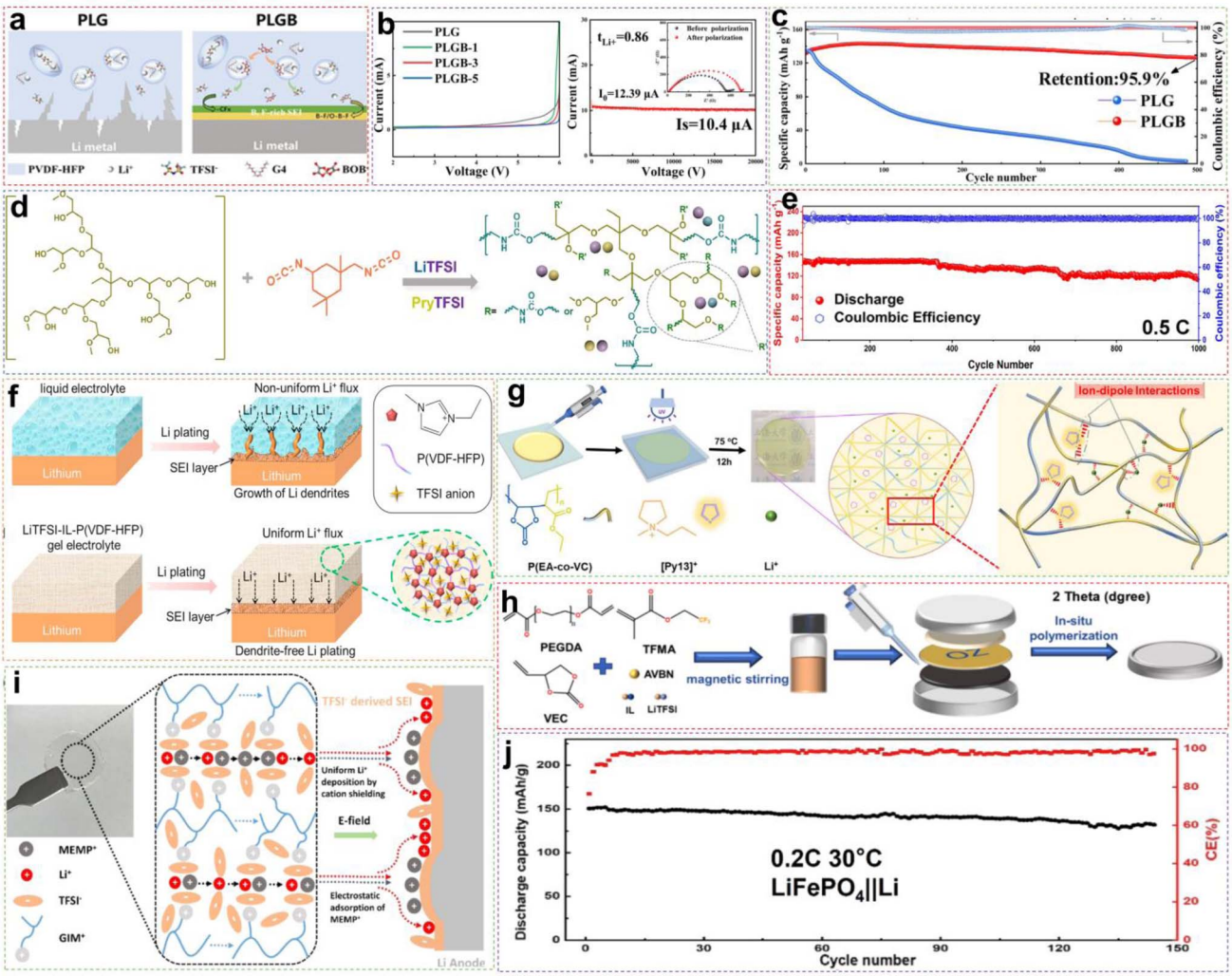

Fig. 11 (a). Schematic representation of Li^+ transport in PLG and PLGB GPE. (b) The LSV measurement of SS/GPE/Li batteries and DC polarization profiles of the symmetrical Li/PLGB/Li battery (the inset shows the impedance spectra before and after polarization). (c) A comparison of the long-term cycle performance between LFP/PLG/Li and LFP/PLGB/Li batteries at 0.5C. Reproduced from ref. 107 with permission. Copyright 2023, The Royal Society of Chemistry. (d) Schematic diagram of the HPU-based electrolyte. (e) Cycling performance of the LFP/HPU1.5-IL1.5/Li cell at 0.5C and RT. Reproduced from ref. 108 with permission. Copyright 2024, Elsevier. (f) Schematic illustration of the electrochemical deposition behavior of lithium metal anodes with liquid organic solution electrolyte and LiTFSI-IL-P(VDF-HFP) gel electrolyte. Reproduced from ref. 109 with permission. Copyright 2018, Elsevier. (g) Schematic diagram for the preparation of CLSPE-IL. Reproduced from ref. 110 with permission. Copyright 2022, Elsevier. (h) *In situ* thermal curing battery assembly flow chart. Reproduced from ref. 111 with permission. Copyright 2024, Elsevier. (i) Schematic illustration of GM-GPE that enhances ionic transport and stabilizes the GPE/Li anode interface. (j) Cycling performances of the LFP/GM-GPE/Li cell at 30 °C. Reproduced from ref. 92 with permission. Copyright 2024, American Chemical Society.

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years, along with their corresponding performances as listed in Table 2.

3.2 IL as a plasticizer for gel polymer electrolytes

Among the various methods used to enhance the ionic conductivities of SPEs, incorporating plasticizers to prepare gel polymer electrolytes (GPEs) is one of the most effective and commonly used approaches.[92,98] Compared with traditional organic solvents, ILs are among the best choices for plasticizers due to their unique advantages, such as extremely low vapor pressure, non-flammability, and good thermal stability.[86,97,101] These GPEs are also known as ionogel electrolytes.[86] On the one hand, ILs facilitate the dissociation of lithium salts through their interactions;[102,103] on the other hand, they weaken the interactions between polymer chains and simultaneously increase the number of amorphous regions, thereby effectively increasing the ionic conductivity of the electrolyte.[104,105] In addition, the special ion channels formed by the ionic liquid in conjunction with other components play a significant role in Li^+ conduction.[106,107] The combined effect of these components enables the electrolyte to exhibit excellent overall performance.

For instance, Yuan *et al.* successfully developed a novel solvate ionic liquid (SIL)-based SPE by incorporating an SIL of $[Li(G_4)_1][TFSI]$ containing the functional additive LiBOB into the PVDF-HFP polymer matrix (named PLGB) (Fig. 11a).[107] The interaction between SIL and PVDF-HFP disrupted the regular arrangement of polymer chains, increasing the proportion of the amorphous phase, which lessens the difficulty of chain movement and ion migration in the PLGB. The addition of LiBOB not only further adjusted the Li^+ coordination environment by competing with $TFSI^-$, but also helped to form the rigid–flexible coupling interface chemistry that buffered the volumetric changes of Li metal during cycling, giving a uniform and dendrite-free Li deposition. Through the synergistic effects of SIL and LiBOB, the solid-state electrolyte displayed a unique solvated structure that promoted Li^+ transport along the polymer matrix. The optimized electrolyte revealed high ionic conductivity (2.18×10^{-3} S cm^{-1}) at RT, high Li^+ transference number (0.86), and outstanding electrochemical stability (up to 5.7 V *vs.* Li/Li^+) (Fig. 11b). The assembled solid-state LFP/PLGB/Li battery delivered a high capacity of 143.2 mA h g^{-1} and capacity retention of 95.9% after 500 cycles at 0.5C (Fig. 11c). Recently, Wang *et al.* reported a hyperbranched polyurethane electrolyte (HPU-IL) by reacting hyperbranched polyether (HPEG) with isophorone diisocyanate (IPDI) in the presence of LiTFSI and 1-*n*-butyl-1-methylpyrrolidinium bis(trifluoromethanesulfonyl)-imide ([Pry][TFSI]) (Fig. 11d).[108] After adding the IL, the X-ray diffraction (XRD) results revealed a new diffuse diffraction peak at 12°, indicating that the IL significantly increased the mobility of the polymer chain segments and interacted with the polymer backbone. This interaction enhanced hydrogen bonding interactions between carbamate and caused microphase separation, which accelerated ion transport. As the battery charged and discharged repetitively, no obvious lithium dendrites were observed on the surface of the LMA. The surface remained smooth and dense, making it clear that the presence of numerous hydrogen bonds in the HPU-IL electrolyte also solves the electrode/electrolyte interfacial contact problem and promotes the formation of a stable SEI at the interface. Owing to the structural advantages of the hyperbranched polyurethane and the plasticizing effect of the IL, the lithium salt dissociation capability, electrochemical stability of the electrolyte, and Li^+ migration capability were significantly improved. The LFP/HPU-IL/Li battery could maintain a discharge capacity of 118 mA h g^{-1} after 1000 stable cycles at 0.5C and RT (Fig. 11e).

Unfortunately, it is worth noting that the excessive introduction of ILs usually brings out an increase of ionic conductivity accompanied by a reduction in mechanical properties, which weakens their ability to resist lithium dendrites.[112] Extensive studies have been conducted to reach a balance between the ionic conductivity and mechanical properties. Chen *et al.* reported the preparation of IL ([EMIM][TFSI]) immobilized GPEs, termed as IL-P(VDF-HFP).[109] As shown in Fig. 11f, the ion–dipole interactions between the imidazolium cation in [EMIM][TFSI] and the polar groups $-CF_x$ in P(VDF-HFP) enable stable and dendrite-free Li^+ plating/stripping. Additionally, a tightly crosslinked gel framework in the polymer matrix takes shape by the incorporation of [EMIM][TFSI] greatly strengthening the mechanical performance and thermal stability. The LiTFSI-IL-P(VDF-HFP) gel electrolyte, with excellent flexibility, also displayed close interfacial contact with electrodes and outstanding self-healing properties. Consequently, the LFP/LiTFSI-IL-P(VDF-HFP)/Li battery demonstrated superior cycling stability and rate performance. Wang *et al.* utilized ethyl acrylate (EA) and vinylene carbonate (VC) as the polymeric monomer, LiTFSI as the lithium salt, and *N*-methyl-*N*-propylpyrrolidine bis (trifluoromethylsulfonyl) imide ($[Py_{13}][TFSI]$) as the additive to prepare a viscoelastic polymer electrolyte with both high ionic conductivity and favorable mechanical properties through UV polymerization (Fig. 11g).[110] The study found that introducing the IL disrupted the crystallization of polymer chains, increasing chain mobility and facilitating Li^+ migration in the electrolyte, which conveniently obtained an ionic conductivity of 2.77×10^{-4} S cm^{-1} at RT. On the other hand, the ion–dipole interactions between the cations in the IL and the oxygen atoms on the copolymer chains enhanced the mechanical properties. So by accommodating the IL content, the polymer electrolyte would possess high tensile strength up to 11.4 MPa and excellent stretchability elongation of 387% at break. Clearly, the resulting LFP/Li battery demonstrated excellent cycling stability at 0.2C, with a discharge capacity of 136.7 mA h g^{-1} after 500 cycles.

Except for utilizing intermolecular interactions to balance ionic conductivity and mechanical properties, forming a reinforcing framework into GPEs is another effective approach. Xu *et al.* successfully prepared a novel polycarbonate-fluorinated solid electrolyte by *in situ* thermal curing method.[111] This involved the use of a polyester separator containing abundant polar functional groups as the reinforcing framework, in conjunction with vinylene carbonate (VEC) and trifluoroethyl methacrylate (TFEMA) as the polymer monomers, poly(ethylene glycol) diacrylate (PEGDA) with a flexible backbone as the

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flexible cross-linking point, and IL ([EMIM][TFSI]) as the plasticizer (Fig. 11h). The introduction of [EMIM][TFSI] promotes dissociation of the lithium salt, generating more mobile Li^+ ions. The strong interaction between $-CF_3$ in TFEMA and $TFSI^-$ competes with Li^+, reducing the coordination ability of $TFSI^-$ and Li^+. This facilitates the coordination and de-coordination of Li^+ among the polymer chains, thereby increasing the Li^+ transference number. Additionally, using a polyester membrane as a reinforcing framework reinforces the mechanical properties of the polymer and its ability to resist lithium dendrite penetration. Through the synergistic effects of these factors, the optimized electrolyte (denoted as 31VPIF/OZ), presented excellent ionic conductivity (3.58×10^{-4} S cm^{-1} at RT), high Li^+ transference number (0.52), and wide electrochemical window (5.4 V). The assembled LFP/31VPIF/OZ/Li battery exhibited 91.5% capacity retention and 99.85% coulombic efficiency after 600 cycles at 0.5C and RT.

ILs play crucial roles as plasticizers in traditional polymer matrices. Because of the strong chemical affinity between PILs and ILs, ILs can be well confined within the PIL, increasing the ionic conductivity and decreasing the risk of leakage.[112] Recently, our research group successfully synthesized a PIL monomer (denoted as [GIM][TFSI]), with polar charges far from the main chain through designing the monomer structure.[92] This structure of the monomer was designed in a way that the imidazole cation away from the main chain increases the free volume of the PIL units, thus facilitating the Li^+ transport rate. Subsequently, by combining this monomer with LiTFSI and the IL plasticizer *N*-methyl-*N*-methoxyethyl-pyrrolidinium bis(trifluoromethylsulfonyl)imide ([MEMP][TFSI]), PIL-based GPE (denoted as GM-GPE) could be obtained by UV light curing. The rich polar charges in GM-GPE effectively promote the dissociation of LiTFSI, and the cationic backbone also can anchor anions to the polymer chains through coulombic interactions (Fig. 11i). The long flexible side chain structure of [GIM][TFSI] and the introduction of the plasticizer will cause the ionic conductivity of GM-GPE to be as high as 4.3×10^{-4} S cm^{-1} at 30 °C. When GM-GPE is applied to LMBs, it is beneficial to generate an SEI film derived from $TFSI^-$ anions, thereby limiting the growth of lithium dendrites. The $MEMP^+$ can migrate to the surface of the lithium metal anode under the influence of the electric field, giving rise to a cationic electrostatic shielding effect, further promoting uniform Li^+ deposition. Thanks to the synergistic effects of these advantages, the LFP/GM-GPE/Li cell displayed a high discharge capacity of 150 mA h g^{-1} at 0.2C, and the capacity retention rate reached 87.6% after 144 cycles at 30 °C (Fig. 11j).

3.3 IL-based composite polymer electrolytes

In 1982, Weston and his colleague Steele first introduced inorganic particles, specifically aluminum oxide (Al_2O_3), into polyethylene oxide (PEO) to prepare solid electrolytes.[113] They found that the conductivity and Li^+ transference number of the electrolyte were slightly impacted with a small amount of Al_2O_3. However, the mechanical properties were significantly improved. Subsequently, various inorganic fillers have been incorporated into polymer systems, leading to the development of CSPEs. The inclusion of inorganic fillers would improve the thermal/electrochemical stability, mechanical properties, ionic conductivity, or ion transference number of polymer electrolytes.[114] Despite these advantages in CSPEs, it's still challenging to modify the organic/inorganic phase interface contact.[115,116] To overcome the interfacial issues, introducing a small amount of IL between the organic and inorganic phases has been considered a convenient and effective strategy. This section focuses on the recent advances in CSPEs with added ILs, emphasizing the role of ILs at the organic/inorganic interface.

3.3.1 Inorganic inert fillers. High specific-surface-area additives such as Al_2O_3,[114] SiO_2 (silica),[117] and TiO_2 (titanium dioxide)[118] as inorganic inert fillers in CSPEs have been well studied. These high specific-surface-area inorganic inert fillers actively interact with the ion pairs and improve their further dissociation, which promotes the number of free Li^+ for participating in conduction through Lewis acid–base interactions.[24,117,119] This process improves the overall ionic conductivity of IL-based CSPEs without compromising the mechanical integrity of the electrolyte, which disrupts the regularity of the polymer chain segments, increasing the amorphous regions and enhancing chain segment mobility.[119] For example, Gandolfo *et al.* prepared a self-supporting flexible membrane composed of an acrylate-based polymer matrix, Al_2O_3 nanoparticles (NPs) and [Pyr_{14}][TFSI] through thermally induced free radical polymerization.[114] The high content of IL and the addition of Al_2O_3 NPs (5 wt%) significantly enhanced the electrochemical performance. The surface acidic sites of Al_2O_3 also improved the dissociation of lithium salts by attracting anions, promoting the release of free lithium ions with notable ionic conductivity (2.5×10^{-4} S cm^{-1} at 20 °C) and wide electrochemical stability window (5.1 V *vs.* Li^+/Li). Compared to the GPE without Al_2O_3, the CSPE with Al_2O_3 exhibited lower polarization, better interface formation capability and higher transference number, which improves the performance of LFP/CSPE/Li batteries, achieving 125 mA h g^{-1} with good CR after 500 cycles at RT.

Similarly, Pan *et al.* prepared a series of CSPEs constructed from PVDF-HFP, [PP_{13}][TFSI], LiTFSI and various inorganic fillers.[118] They found that the mechanical properties of the CSPE were improved with the addition of 5 wt% TiO_2 particles, which formed a dense electrolyte membrane by increasing the amorphous phase of PVDF-HFP and filling the pores of the CSPE. The mechanical properties would be improved by the addition of inert fillers; however, the ionic conductivity might be significantly reduced when large amounts of inert fillers were added, owing to the aggregation of nanoparticles. Therefore, the content of nanoparticles as inert fillers is typically limited to less than 10 wt% of the total CSPE in most previous studies. Recently, Kim *et al.* synthesized vinyl mesoporous silica nanoparticles (VMSNs) as reinforcing fillers by introducing vinyl groups on the surface of mesoporous silica nanoparticles (MSNs) *via* the surface grafting method.[117] Compared to physically dispersed CSPEs containing MSNs, functionalized VMSNs can chemically crosslink with poly(ethylene glycol) (PEG) oligomers in the polymer matrix, enhancing the dispersion of

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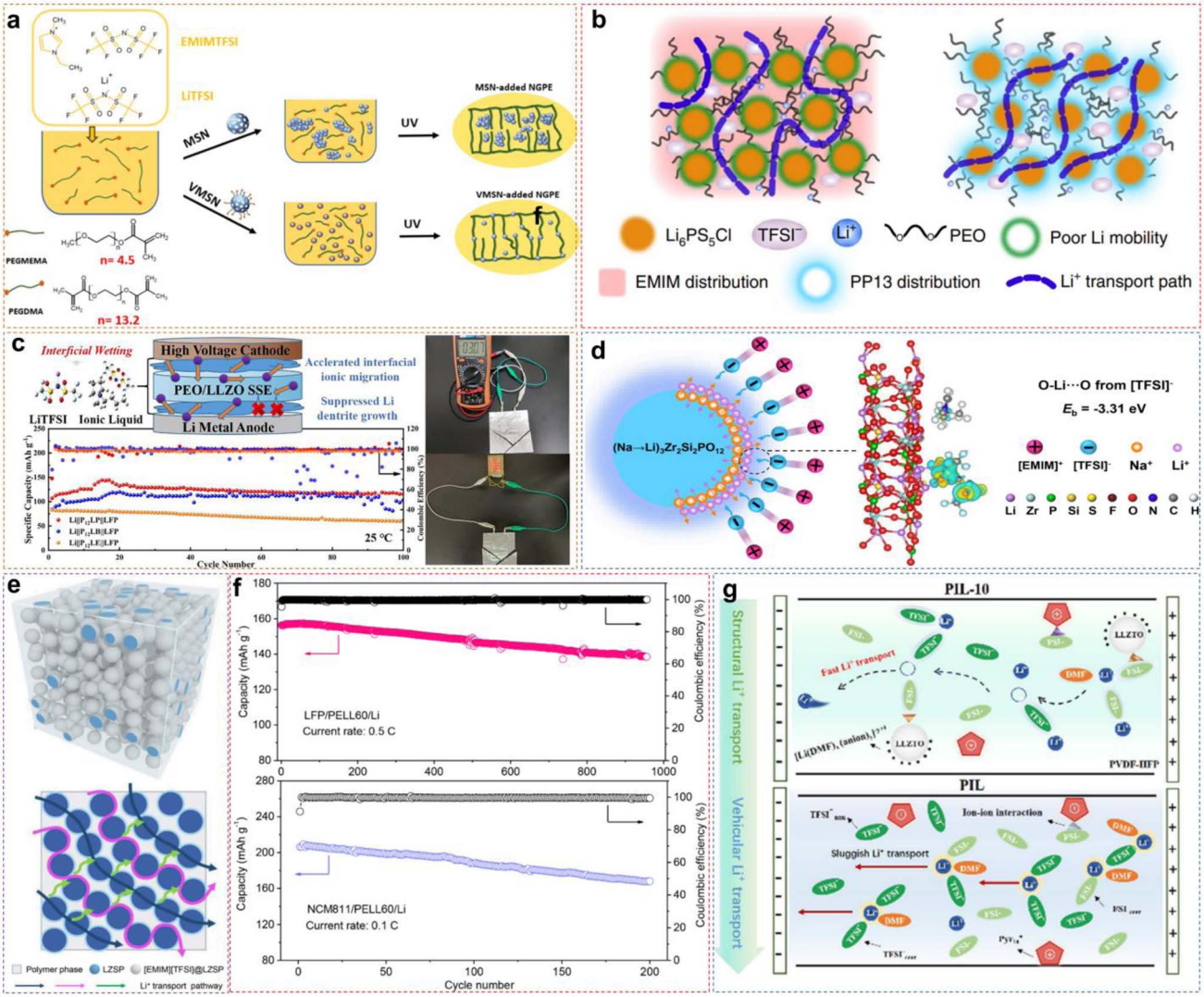

Fig. 12 (a) Synthetic schematic diagram of nanohybrid gel polymer electrolytes (MSN-based NGPE and VMSN-based NGPE). Reproduced from ref. 117 with permission. Copyright 2023, American Chemical Society. (b) Proposed mechanism for Li^+ diffusion in CSPEs with [EMIM][TFSI] and [PP_{13}][TFSI] IL additives. Reproduced from ref. 120 with permission. Copyright 2022, Springer Nature. (c) Schematic diagram of CSPEs and pouch-type Li/PEO/LLZO@IL/LFP cells at 0.1C at RT. Reproduced from ref. 116 with permission. Copyright 2022, Elsevier. (d) Schematic diagram and DFT calculations on the interactions between LZSP surfaces and the [EMIM][TFSI] PCIL. (e) Schematic of the Li^+ diffusion mechanism for PELL60. (f) Long-time cycling performance of LFP/PELL60/Li batteries at 0.5C and long-term cycling performance of NCM811/PELL60/Li batteries at 0.1C and RT. Reproduced from ref. 121 with permission. Copyright 2024, American Chemical Society. (g) Schematic illustrations of Li^+ solvation structure and transformed transport mechanism in PIL-10 and PIL. Reproduced from ref. 122 with permission. Copyright 2024, Elsevier.

mesoporous silica nanoparticles (Fig. 12a). This offers CSPEs higher mechanical properties (1.1×10^6 Pa) and higher ionic conductivity (1×10^{-4} S cm^{-1}) at RT.

3.3.2 Inorganic active fillers. The introduction of inorganic inert fillers can reduce the crystallinity of polymers to increase ionic conductivity. However, inert fillers do not participate directly in the construction of ion transport pathways.[115] Recently, CSPEs with inorganic active fillers and ILs as additives have been extensively investigated. The addition of incorporating inorganic active fillers (*e.g.*, $Li_{1+x}Al_xTi_{2-x}(PO_4)_3$ (LATP)[115] and $Li_{6.4}La_3Zr_{1.4}Ta_{0.6}O_{12}$ (LLZTO)[123]) is aimed to create more pathways for Li^+ transport, thus improving the overall conductivity of CSPEs.[115,119] The addition of a slight amount of ILs does not diminish the mechanical strength and flexibility of the CSPEs. Instead, it significantly improves the interface contact issues. However, the complex mechanism of Li^+ transport at the organic/inorganic interfaces and the specific role of ILs require further investigation.[116] Liu *et al.* investigated the interfacial structure and transport mechanism of Li^+ at the interface of CSPEs using multinuclear solid-state nuclear magnetic resonance (NMR).[120] They found that an inert interface formed between the organic phase PEO and the inorganic phase Li_6PS_5Cl in the CSPE, resulting in very slow diffusion of Li^+ at the interface. To improve the interface and facilitate Li^+ diffusion, they respectively introduced an imidazolium-based IL ([EMIM][TFSI]), which has high miscibility with PEO, and poor

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miscibility with piperidinium-based IL ([PP_{13}][TFSI]) in the CSPE. The results showed that [EMIM][TFSI] primarily existed in PEO, reducing the crystallinity of PEO. In contrast, the poorly miscible [PP_{13}][TFSI] was forced to remain at the interface of PEO and Li_6PS_5Cl, wetting the polymer–inorganic interface and acting as a bridge for Li^+ transport, thus increasing the overall ionic conductivity (2.47×10^{-4} S cm^{-1}) of the CSPE (Fig. 12b).

The introduction of ILs can result in higher ionic conductivity and more thorough interfacial contact. However, excessive amounts of ILs can pose safety issues and increase the overall cost of the batteries. As shown in Fig. 12c, Yu *et al.* optimized the IL content in PEO/$Li_7La_3Zr_2O_{12}$ (LLZO) CSPEs and systematically studied the impact of IL cations on the interfacial behavior at the lithium anode side.[116] With increasing IL content, the ionic conductivity and thermal stability of the CSPE were enhanced, and the EEI contact was simultaneously increased. However, when the IL content exceeded 17.5 wt%, the leakage of IL from the CSPE was observed, posing safety risks. Additionally, ILs directly participated in the SEI formation at the CSPE/Li interface, where ILs based on pyrrolidinium cations ($[Py_{14}]^+$) showed significantly more compatibility with the LMA, compared to ILs based on imidazolium cations ($[BMIM]^+$ and $[EMIM]^+$). The addition of [Py_{14}][TFSI] resulted in the LiF and Li_3N-riched SEI layer at the CSPE/Li interface, which facilitated Li^+ conduction and suppressed lithium dendrite growth. The assembled pouch-type solid-state LFP/CSPE/Li cell achieved a specific capacity of 120 mA h g^{-1} and CE of more than 99% after 100 cycles at 0.1C.

The uniform dispersion of inorganic particles within the polymer matrix is crucial to achieve excellent overall performance of CSPEs. Like inorganic inert fillers, it is difficult to uniformly disperse the fillers within the polymer matrix when the content of the active filler is increased (beyond 10 wt%). Zhu *et al.* proposed using polymer-compatible ionic liquids (PCILs) to address the interfacial issues between the fillers and the polymer matrix.[121] Using [EMIM][TFSI] as the liquid carrier, the $Li_3Zr_2Si_2PO_{12}$ (LZSP) was synthesized *via* the Na^+/Li^+ cationic exchange method. During the process, the cations concentrated on the filler surface and interacted with the adjacent PCIL, increasing the repulsive force and distance between particles and effectively solving the aggregation problem (Fig. 12d). The approximately 10 nm thick [EMIM][TFSI] coating uniformly covered the LZSP particle surfaces. The prepared PELL60 CSPE consisted of 30 wt% PVDF, 60 wt% [EMIM][TFSI]@LZSP, and 10 wt% LiTFSI. Subsequently, solid-state NMR revealed that in the CSPE composed of unmodified LZSP phases, the main conduction pathway for Li^+ was through the LZSP phase. In contrast, PELL60 CSPE not only contributes to the highly dispersed LZSP powder, which builds up the Li^+ conduction pathways in the native LZSP, but also facilitates pathways at the [EMIM][TFSI]@LZSP-PVDF interface and the intermediate spaces in PELL60 CSPE (Fig. 12e). As a result of the interconnected Li^+ transport pathways established by [EMIM][TFSI] PCIL, the resulting PELL60 CSPE achieved a perfect combination of high ionic conductivity (8.3×10^{-4}S cm^{-1}), high Li^+ transference number (0.81), excellent flexibility, and strong mechanical strength. The PELL60 CSPE exhibits decent cycling performance in both LFP and NCM811 batteries (Fig. 12f), as well as a high safety and high energy density of 424.9 W h kg^{-1} (excluding packing materials) in NCM811/PELL60/Li solid-state Li metal pouch batteries.

Recently, Lin *et al.* were the first to reveal the influence of inorganic active fillers on the solvation structure in IL-based CSPEs and their role in forming SEI layers on Li metal.[122] IL containing $FSI^-/TFSI^-$ anions and the LLZTO active filler were added to the PVDF-HFP matrix to prepare the CSPE. Detailed studies on the Li^+ transport mechanism showed that LLZTO selectively anchored the $FSI^-/TFSI^-$ anions, thereby altering the migration behavior of Li^+ and the ratio of $TFSI^-/FSI^-$ anions in the solvation shell. Adjusting the local environment aids in competing for coordination with $TFSI^-$ anions in the solvent structure dominated by FSI^- anions, thereby forming a more stable interfacial chemistry (Fig. 12g). In the PIL-10 CSPE, the change in ionic environment enhances conductivity (1.24×10^{-3} S cm^{-1}) and Li^+ transference number (0.42) because of the preferable shift from the vehicular to structural Li^+ transport. Additionally, the synergistic coordination of the solvent and the adjustment of the EEI offer the LFP/CSPE/Li stable battery cycling, achieving 95.4% CR after 500 cycles at 1C and RT. In the future, the design of composite electrolytes with high ionic conductivity, wide electrochemical window, high mechanical strength, and good interfacial contact and compatibility is a major research focus.

4 Electrolyte/electrode interface

The interface consists of two parallel sheets of excess charge, equal in magnitude but opposite in sign, commonly referred to as an "electric double layer". Ideally, this interface would have zero thickness, and no irreversible reactions would occur between the electrode and the electrolyte (Fig. 13a).[124] In reality, metallic Li, with its highly negative potential of −3.04 V *vs.* the standard hydrogen electrode, is extremely reactive and interacts with nearly any substance it contacts. When the electrode potential exceeds the electrochemical stability window of the electrolyte, a stable new layer forms on the electrode surface, known as the "interphase". This is termed the SEI and CEI on the anode and cathode side, respectively (Fig. 13b).[125] Both SEI and CEI are crucial for maintaining stable battery cycling. Ideally, SEI and CEI are electronic insulators and ionic conductors, preventing electrons from leaking out of the electrode and reacting with the electrolyte while allowing efficient lithium-ion transport. However, natural SEI and CEI typically exhibit low ionic conductivity and poor mechanical strength, leading to electrolyte penetration through the electrode/electrolyte interface (EEI) and continuous severe side reactions.

Although ionic liquids (ILs), as either liquid or solid electrolytes, can stabilize LMBs during cycling, their high cost remains a significant limitation. Therefore, researchers are increasingly focused on how to achieve substantial performance improvements in LMBs with low doses of ILs. This section will explore three main approaches: using ILs as additives, pretreating electrodes with ILs, and employing ILs as interface wetting agents. The low-dose ILs can introduce a rich variety of

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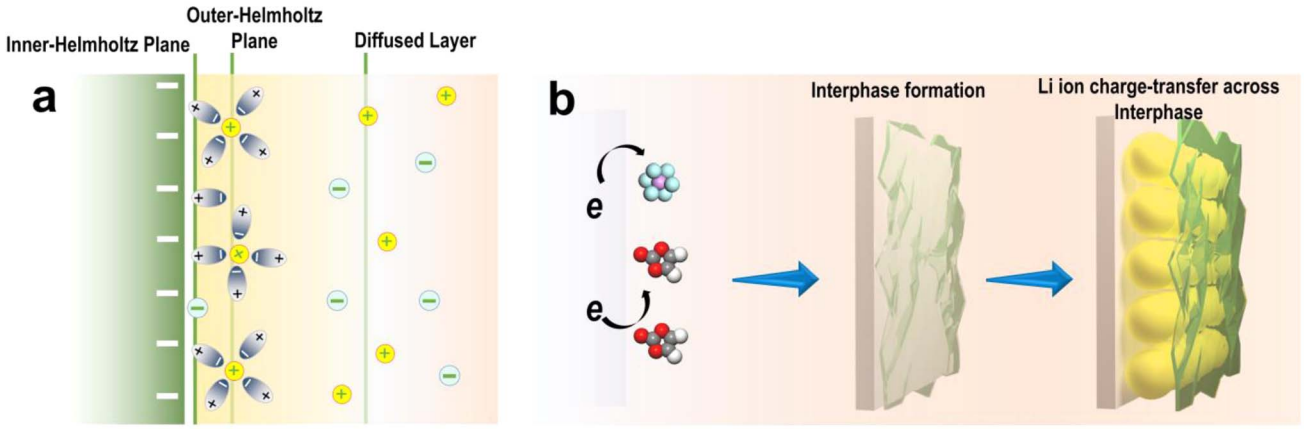

Fig. 13 (a) Schematic of electric double layer. (b) Illustration of the interphase formation and Li ion charge transfer process.

organic cations and inorganic anions into ether- or ester-based electrolytes. The Li dendrite protrusions typically exhibit higher electric potential, attracting organic cations that adsorb onto these sites, creating an electrostatic shielding effect to prevent excessive dendrite growth. Inorganic anions, on the other hand, become part of the Li^+ ion solvation structure and decompose at the electrode surface, introducing inorganic components that stabilize the EEI. Moreover, using ILs for pre-treating electrodes is a cost-effective and scalable strategy. This process essentially constructs an artificial EEI. For instance, PILs can be applied to coat graphite or LMAs, reducing side reactions at the interface. In solid-state electrolyte applications, low-dose ionic liquids can enhance ionic conductivity at the interface, effectively lowering interfacial impedance. In summary, using the non-electrolytic

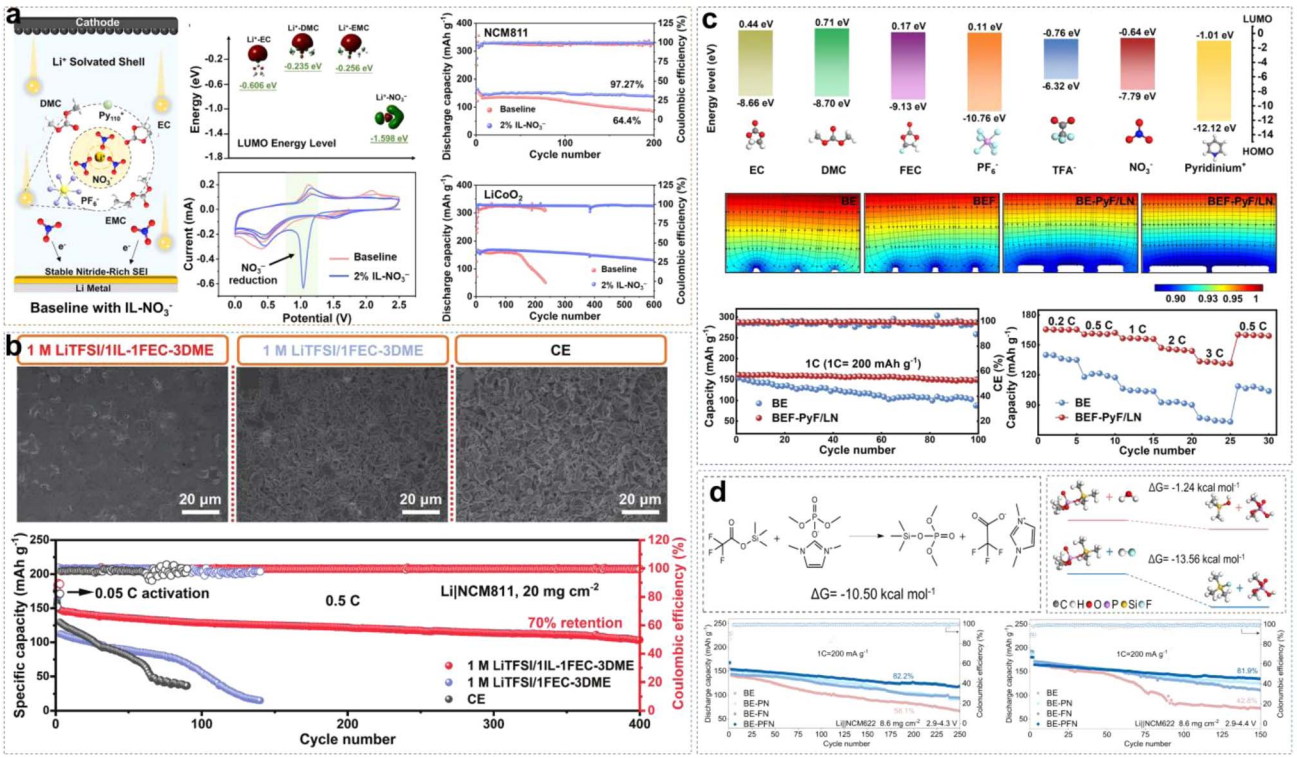

Fig. 14 (a) Solvation structure of 2% IL-NO_3^- electrolyte; LUMO energy level of Li^+-complexes; CV curves of Li/Cu cells for the baseline electrolyte and 2% IL-NO_3^- electrolyte; cycling performance of Li/NCM811 and Li/LCO cells. Reprinted from ref. 128 with permission. Copyright 2023, Elsevier. (b) SEM images of the cycled LMA in different electrolytes and corresponding cycling performance of Li/NCM811 cells. Reprinted from ref. 129 with permission. Copyright 2022, Wiley. (c) Molecular orbital energies of solvents and additives calculated by DFT; finite element analysis of Li deposition morphology simulated by COMSOL Multiphysics; cycling and rate performance of Li/LCO cells. Reprinted from ref. 130 with permission. Copyright 2022, Elsevier. (d) The reaction between DIDP and TMSF and the reaction paths of DTMSP and H_2O/HF; long-term cycling performance in Li/NCM622 cells. Reprinted from ref. 131 with permission. Copyright 2024, Wiley.

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functions of ILs to improve EEI stability marks a significant advancement in the application of ILs in LMBs.

4.1 SEI/CEI-forming additives

Lithium nitrate ($LiNO_3$) has proven to be an effective additive for forming a stable SEI, but it has poor solubility in carbonate-based electrolytes. This solubility can be enhanced using cosolvents like dimethyl sulfoxide (DMSO) and sulfolane (SL).[126,127] However, these cosolvents can compromise SEI stability, negatively impacting battery cycling performance. Therefore, Ma *et al.* designed 1-decyl-1-methylpyrrolidinium nitrate ($Py_{110}{}^{+}NO_3{}^{-}$), with weaker binding energy as an electrolyte additive. $NO_3{}^{-}$ can incorporate into the Li^+ solvation structure, and the low LUMO orbital of Li^+–$NO_3{}^{-}$ facilitates its reduction on LMA surfaces, forming a stable SEI layer (Fig. 14a).[128] Using 2% IL-$NO_3{}^{-}$, Li/NCM811 cells retained 97.27% capacity after 200 cycles at 1C rate, and Li/$LiCoO_2$ cells retained 80% capacity retention after 600 cycles. Huang *et al.* employed [EMIm][NO_3] IL as a solvent component, achieving dense and smooth lithium deposition with Li^+-coordinated $NO_3{}^{-}$ solvation structures (Fig. 14b).[129] With a mass loading of 20 mg cm^{-2}, Li/NCM811 cells retained 70% capacity retention after 400 cycles at the 0.5C rate. Fang *et al.* introduced pyridinium trifluoroacetate (PyTFA) as an additive into commercial carbonate electrolytes.[130] The strong coordination between the carbonyl (C=O) in trifluoroacetate anions (TFA^-) and Li^+ facilitated the dissolution of $LiNO_3$ in the carbonate electrolyte, creating an anion-rich formation of an inorganic-rich SEI, inducing uniform Li^+ deposition (Fig. 14c). Li/NCM523 full cells using BEF-PyF/LN demonstrated excellent cycling stability with 92.9% CR after 100 cycles and a CE of 99.3%, while maintaining a capacity of 133.5 mA h g^{-1} at 3C. Single additives often struggle to meet the diverse electrochemical performance requirements of LMBs. Therefore, the synthesis and introduction of multifunctional additives are urgently needed for high-performance batteries. Gao *et al.* discovered that 1,3-dimethylimidazole dimethylphosphate (DIDP) and trimethylsilyl trifluoroacetate (TMSF) can undergo *in situ* transesterification in carbonate electrolytes to generate dimethyltrimethylsilyl phosphate (DTMSP) and 1,3-dimethylimidazole trifluoroacetate (DITFA) as multifunctional additives for LMBs (Fig. 14d).[131] Due to the Si–O groups, DTMSP effectively removes H_2O and HF, enhancing the moisture resistance of the electrolyte and contributing to the stability of the cathode.

4.2 Interface pretreatment and wetting

Utilizing ILs to spontaneously form an SEI on electrode surfaces can significantly enhance electrode cycling stability. Wang *et al.*

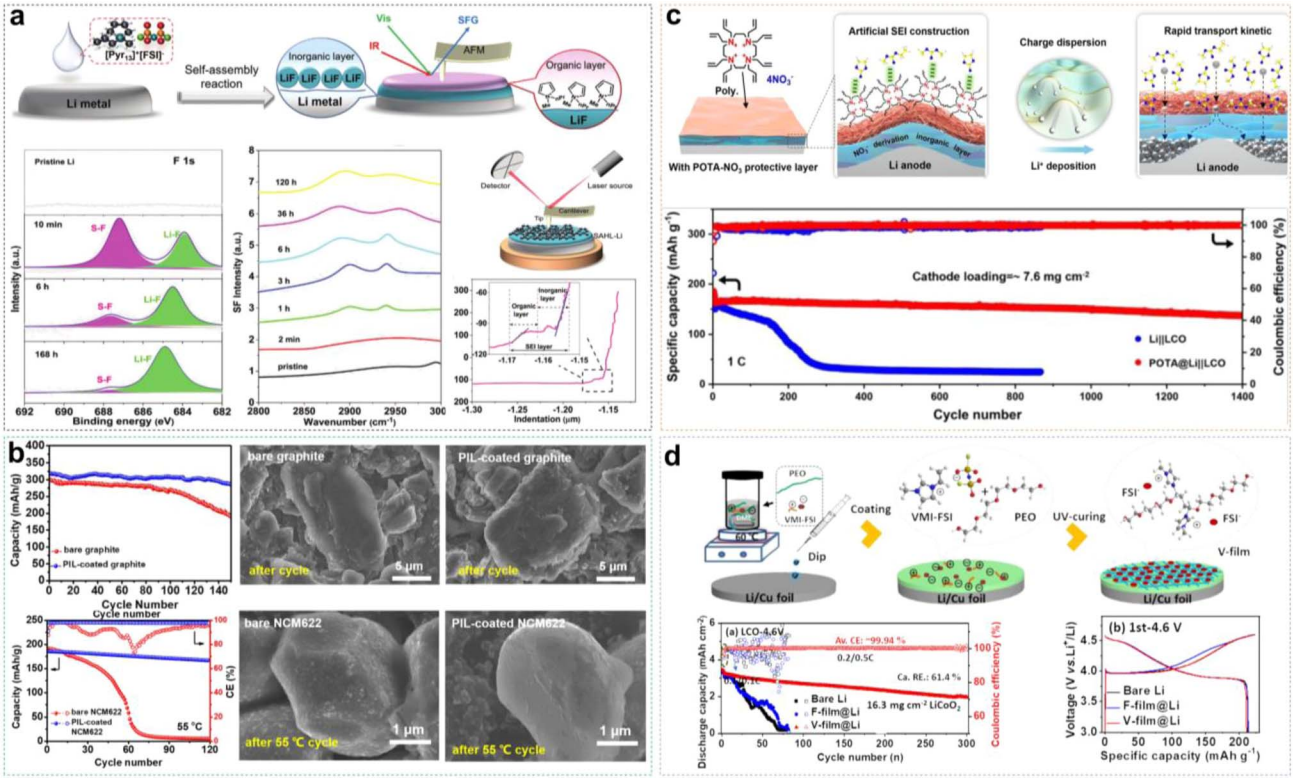

Fig. 15 (a) Schematic illustration of artificial SEI formation on LMAs and corresponding evolution process probed by XPS, SFG and AFM. Reprinted from ref. 132 with permission. Copyright 2020, Wiley. (b) Cycling performance of Li/Gr and Li/NCM622 cells with PIL coating layer, and corresponding SEM images of cycled electrodes. Reprinted from ref. 133 with permission. Copyright 2022, Elsevier. (c) Schematic illustration of Li^+ plating modulation by the POTA-NO_3 protective layer and Li/LCO cycling performance. Reprinted from ref. 134 with permission. Copyright 2024, Elsevier. (d) Schematic illustrating the preparation of V-film, and cycling performance of Li/$LiCoO_2$ full cells at cutoff voltages of 4.6 V. Reprinted from ref. 135 with permission. Copyright 2024, Wiley.

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demonstrated that immersing LMA in $Pyr_{13}FSI$ leads to the *in situ* growth of an ordered organic/inorganic hybrid SEI layer, where the reactive FSI^- anions reacts with metallic Li to form LiF.[132] XPS and sum frequency generation (SFG) spectroscopy confirmed the dynamic formation process of LiF. The force curves of the atomic force microscopy (AFM) images showed a two-stage change: an initial slow increase followed by a plateau and then a sharp increase, indicating the mechanical differences between the organic and inorganic layers in the SEI (Fig. 15a). In addition to spontaneous formation, SEI can also be artificially constructed. In our previous work, we designed a cationic polymeric liquid (PIL) as an artificial SEI for high-voltage NCM622 cathodes and graphite anodes, aiming to suppress electrolyte decomposition on electrode surfaces.[133] This PIL film, with its high mechanical modulus and excellent electrochemical and chemical stability, effectively inhibited interfacial side reactions. Its abundant cationic groups had strong coulombic interactions with PF_6^- anions, promoting lithium-ion conduction within the SEI. As shown in Fig. 15b, the PIL coating layer significantly improved the cycling stability of Li/Gr and Li/NCM622 cells. SEM images of cycled electrodes revealed that unmodified graphite particles had fuzzy boundaries and were covered by a thick coating due to severe EEI side reactions, while the morphology of PIL-coated graphite particles remained largely unchanged. Additionally, unmodified NCM622 particles exhibited a thick layer of degradation products on their surface after high temperature cycling. In contrast, the surface of the PIL-coated NCM622 particles remained uniformly smooth. Yu *et al.* developed a high-charge-density cationic polymer, poly(octaallyltetraazacyclodecane nitrate) (POTA-NO_3), as an artificial SEI (Fig. 15c).[134] The electrostatic shielding effect of OTA_4^+ effectively regulated lithium-ion deposition paths and facilitated the desolvation process. NO_3^- generated a stable inorganic SEI layer, preventing the decomposition of the polymer cation layer and electrolyte. POTA@Li/LCO cells demonstrated over 80% CR after 1400 cycles with a CE exceeding 98%. Qin *et al.* cross-linked ionic liquid 1-vinyl-3-methylimidazolium bis(fluorosulfonyl)imide (VMI-FSI) with polyethylene oxide (PEO) to form a self-healing membrane.[135] FSI^- was chemically decomposed into LiF and Li_3N, aiding SEI formation on lithium and repairing SEI film cracks caused by lithium dendrite tearing (Fig. 15d).

ILs can also be used to wet solid-state interfaces and reduce interfacial resistance between solid electrolytes and cathodes. Im *et al.* prepared quasi-solid-state LMBs using Ag-coated $Li_{6.4}La_3Zr_{1.7}Ta_{0.3}O_{12}$ powder (LLZTO), an Ag/C composite interlayer, and NCM333 cathodes, wetting the cathode and LLZTO@Ag/C with a 2 M LiFSI/$Pyr_{13}FSI$ additive.[136] Shen *et al.* significantly enhanced the room-temperature performance of solid-state LMBs using a pyrrole-based IL (1-butyl-1-methyl pyrrolidinium bis(trifluoromethanesulfonyl)-imide) (BMP-IL) interlayer.[137] The IL interlayer greatly facilitated lithium-ion conduction and reduced interfacial impedance by 400 times. The Li/LCO cell with the BMP-IL interlayer showed a high specific capacity of 122 mA h g^{-1} and an ultrahigh CR of 96% after 100 cycles.

5 Summary and perspectives

The lithium metal anode is fundamental to achieving high-energy-density lithium batteries. Ionic liquids, with their unique properties, such as excellent thermal stability, high ionic conductivity, and low flammability have further propelled the development of LMBs. In this review, we have systematically discussed the latest developments of ILEs and PILEs in LMBs and summarized strategies for achieving stable cycling by regulating solvation and interfacial chemistry in different IL-based electrolytes. From the above directions, we can conclude that the structural diversity and adjustability of ILs allow the precise molecular design and facilitate collaboration with other components to modulate solvation structures and interfacial chemistry, thereby adapting to various application scenarios. Although certain progress in IL-based electrolytes has been achieved, there remain many challenges in the following aspects.

Firstly, optimizing the synthesis processes and designing stable and cost-effective ILs are the primary challenges facing the commercialization of IL-based electrolytes. From an economic perspective, directly replacing existing electrolytes by IL-based ones is expected to significantly increase battery costs. This is because the preparation of ILs typically requires large amounts of organic solvents, and the purification processes are complex. Besides, the current design process for ILs resembles the mixing of cocktails, necessitating tedious and extensive experimental screening, which limits the rapid development of new ILs.

Secondly, the underlying mechanisms of ILs in LMBs require further investigation, with complexity research involving electrochemistry and interfacial chemistry. In IL-based electrolytes, the solvation structure of Li^+ significantly influences the

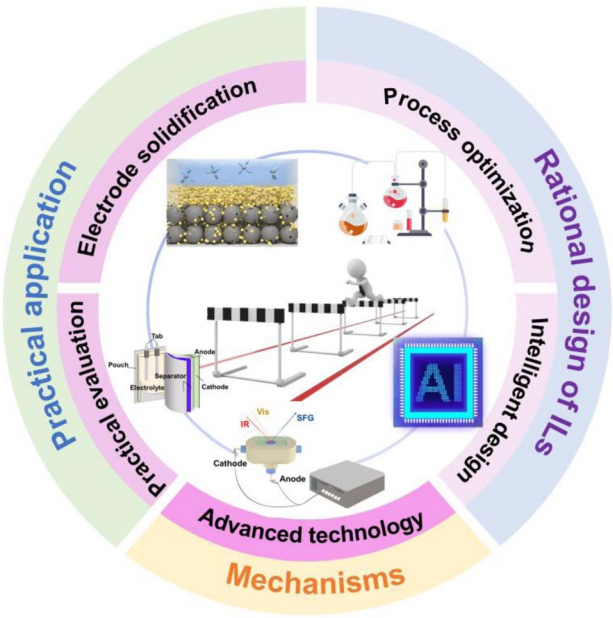

Fig. 16 Prospects of ILs for future development.

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properties of the EEI, which is directly associated with the electrochemical performance of LMBs. Currently, characterization techniques are widely applied, including Raman spectroscopy, Fourier-transform infrared spectroscopy (FTIR), and nuclear magnetic resonance (NMR). However, it remains challenging to directly determine the influence of solvent molecules on solvation chemistry, the impact of solvation chemistry on interfacial chemistry, and ultimately, the effect of these factors on electrochemical performance.

Finally, to broaden the practical applications of ILs, further development at the cell device level is needed to address issues such as overall ion conduction and interfacial contact. Additionally, most evaluations of IL-based electrolytes are conducted in coin cells with excessive lithium (*e.g.*, using lithium foil thickness >300 μm). Therefore, the reliability of IL-based electrolytes under practical conditions still requires further assessment.

For facilitating the practical development of IL-based electrolytes, our views on future research directions are listed below (Fig. 16).

5.1 Rational design of ILs

In the synthesis and purification of ILs, more streamlined experimental routes are required to reduce production costs. Emerging technologies such as microwave irradiation and ultrasound-assisted reaction techniques present promising solutions. Besides, the development of efficient new catalysts can significantly improve reaction rates and selectivity, thereby simplifying synthesis procedures and minimizing byproduct formation.

In terms of IL design, leveraging computer science can reduce the trial-and-error costs associated with traditional design methods. By extracting useful patterns from large datasets, computer science enables the rapid design of new ILs, significantly accelerating the development cycle. For instance, computers can be used to combine various types of cations and anions, with machine learning (ML) in artificial intelligence (AI) to predict IL properties such as melting point, viscosity, and conductivity. Deep learning (a subset of ML) can be performed to screen and evaluate ILs with the liquid phase at room temperature and ionic conductivities larger than 5 mS cm^{-1}. The electrochemical window values calculated from the energy levels of ILs can serve as screening criteria to identify ILEs suitable for different high-voltage cathode materials. All results from this process are stored in a database, providing sufficient data support for ML to continue optimizing design schemes. Additionally, different functional groups can be introduced into cations or anions during the screening process to functionalize ILs for better performance. Furthermore, by screening, identifying, and simulating the interactions of various ILs and other electrolyte components (including salts, solvents, and diluents), the synergistic effects of multiple components can be effectively balanced, and electrolyte formulations can be optimized. This approach offers both theoretical and practical guidance for the efficient and intelligent development of new IL-based electrolytes in the future.

5.2 Profound comprehension of mechanisms

Developing multi-scale *in situ* characterization techniques is essential for a comprehensive understanding of actual

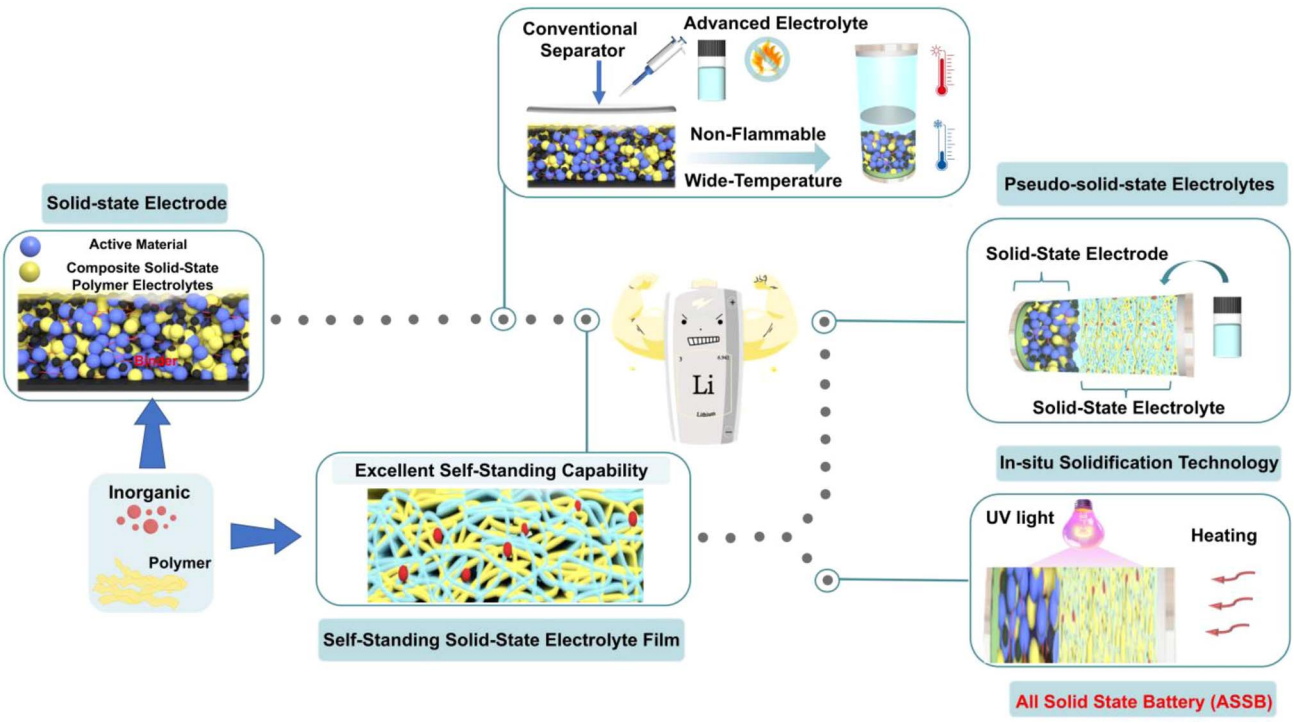

Fig. 17 Schematic illustration of solidified electrodes for applications in liquid and solid-state electrolytes.

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electrolyte behaviors about solvation/desolvation and interfacial interactions during charge/discharge. For example, *in situ* magnetic resonance imaging (MRI), *in situ* FTIR, and super-resolution electrochemical microscopy can provide insights into the dynamic evolution of Li^+-solvent/anion solvation environments and interfacial structures. This further elucidates the structure–property relationships of ILs and interfacial microchemical structures, thereby offering valuable guidance to design and synthesize more effective ILs. In addition to employing advanced characterization techniques, computational methods including DFT and MD have been utilized to study the solvation structures and interfacial chemistry of electrolytes. However, simplified models or approximate algorithms may not accurately describe the properties of different electrolytes. Therefore, precise and efficient new computational methods must be developed for the dynamic structures in complex systems, constructing cross-scale theoretical models that span from microscopic to mesoscopic levels. Researchers must also establish a relationship between theoretical calculations and experimental characterization results to ensure the reliability and validation of theoretical models.

5.3 Practical application

Based on the unique physicochemical properties of ILs and PILs, we propose a novel concept of solid-state electrodes. In particular, as illustrated in Fig. 17, these solid-state electrodes are composed of active materials, inorganic nanoparticles, PILs, and ILEs. The structural characteristic of solid-state electrodes is that PILs serve as the main filling matrix, while ILEs facilitate wetting the polymer/electrode interface and act as a bridge for Li^+ transport. To further enhance its performance, nano-inorganic fillers can also be incorporated into polymer electrolytes to construct additional ion transport pathways. The solid-state electrodes can: (i) be paired with traditional separators and liquid electrolytes to achieve high specific energy batteries with a wide temperature range and high rate capabilities; (ii) be combined with self-supporting solid electrolyte film, with a minimal amount of liquid electrolyte added to aid interfacial wetting, to realize pseudo-solid-state batteries; and (iii) ultimately achieve true all-solid-state batteries through *in situ* solidification techniques such as heating or lighting exposure in conjunction with solid electrolytes. This innovative solution provides greater flexibility in the electrochemical system design. The simple and convenient preparation process is compatible with existing lithium-ion battery production systems, reducing system transition costs and providing a new strategy for advancing high-energy-density batteries.

Additionally, to facilitate the commercialization of IL-based electrolytes, it is essential to evaluate these electrolytes under conditions involving high current stripping/plating processes or within high-loading pouch cells. For example, to achieve higher energy densities, such as 500 W h kg^{-1}, the areal stripping/plating capacity must exceed 3 mA h cm^{-2}, and the charging current density should increase to 3.0 mA cm^{-2}. This requirement implies that high-energy-density batteries need to be fully charged within one hour.[138] Consequently, many reported strategies must be re-evaluated to determine whether IL-based electrolytes can meet these performance thresholds and achieve satisfactory cycling performance. Furthermore, energy density should be assessed based on cell-level packaging rather than relying on rough estimates of the active materials. Due to complex interfacial interaction, the transition from laboratory-grade batteries to applications will be a huge challenge. Therefore, it is crucial to establish the targeted practical evaluation system and assessment standards to support the design of IL-based electrolytes that are suitable for high-energy-density battery applications.

In summary, future research should focus on IL-based electrolytes, including structural design, profound understanding of mechanisms, and practical applications to promote their widespread development in high-energy-density LMBs, establishing a solid foundation for the next generation of efficient and environmentally friendly battery technologies.

Data availability

Data availability is not applicable to this article as no new data were created or analyzed in this review.

Author contributions

X. Wu and J. Xu conceived the idea and theme. H. Tu, K. Peng and J. Xue wrote the draft manuscript and contributed equally to this work. J. Zhao, Y. Guo, S. Lu, Z. Wang, L. Chen, H. Li, X. Wu and J. Xu edited and polished the manuscript. J. Xu and X. Wu supervised the project. All the authors contributed to the discussion and writing of this review.

Conflicts of interest

There are no conflicts of interest to declare.

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Notes and references

1 M. V. Reddy, A. Mauger, C. M. Julien, A. Paolella and K. Zaghib, *Materials*, 2020, **13**, 1884.
2 J. Zhang, H. Zhang, S. Weng, R. Li, D. Lu, T. Deng, S. Zhang, L. Lv, J. Qi, X. Xiao, L. Fan, S. Geng, F. Wang, L. Chen, M. Noked, X. Wang and X. Fan, *Nat. Commun.*, 2023, **14**, 2211.
3 D. Lu, R. Li, M. M. Rahman, P. Yu, L. Lv, S. Yang, Y. Huang, C. Sun, S. Zhang, H. Zhang, J. Zhang, X. Xiao, T. Deng, L. Fan, L. Chen, J. Wang, E. Hu, C. Wang and X. Fan, *Nature*, 2024, **627**, 101–107.

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4 X. Huang, R. Li, C. Sun, H. Zhang, S. Zhang, L. Lv, Y. Huang, L. Fan, L. Chen, M. Noked and X. Fan, *ACS Energy Lett.*, 2022, **7**, 3947–3957.
5 L. Li, M. Wang, J. Wang, F. Ye, S. Wang, Y. Xu, J. Liu, G. Xu, Y. Zhang, Y. Zhang, C. Yan, N. V. Medhekar, M. Liu and Y. Zhang, *J. Mater. Chem. A*, 2020, **8**, 8033–8040.
6 H. Tu, L. Li, Y. Hu, Y. Zhang, Y. Wang, W. Huang, Z. Ren, H. Lin and M. Liu, *Chem. Eng. J.*, 2022, **434**, 134647.
7 L. Li, H. Tu, J. Wang, M. Wang, W. Li, X. Li, F. Ye, Q. Guan, F. Zhu, Y. Zhang, Y. Hu, C. Yan, H. Lin and M. Liu, *Adv. Funct. Mater.*, 2023, **33**, 2212499.
8 H. Tu, Z. He, A. Sun, F. Mushtaq, L. Li, Z. Wang, Y. Kong, R. Huang, H. Lin, W. Li, F. Ye, P. Xue and M. Liu, *Nano Lett.*, 2024, **24**, 5714–5721.
9 J. Zhou, B. Hao, M. Peng, L. Zhang, H. Ji, J. Liu, W. Ling, C. Yan and T. Qian, *Adv. Energy Mater.*, 2023, **13**, 2204174.
10 H. Ji, Z. Wang, Y. Sun, Y. Zhou, S. Li, J. Zhou, T. Qian and C. Yan, *Adv. Mater.*, 2023, **35**, 2208590.
11 G. Liang, V. K. Peterson, K. W. See, Z. Guo and W. K. Pang, *J. Mater. Chem. A*, 2020, **8**, 15373–15398.
12 M. S. Kim, Z. Zhang, P. E. Rudnicki, Z. Yu, J. Wang, H. Wang, S. T. Oyakhire, Y. Chen, S. C. Kim, W. Zhang, D. T. Boyle, X. Kong, R. Xu, Z. Huang, W. Huang, S. F. Bent, L. W. Wang, J. Qin, Z. Bao and Y. Cui, *Nat. Mater.*, 2022, **21**, 445–454.
13 M. Ma, R. Huang, M. Ling, Y. Hu and H. Pan, *Interdiscip. Mater.*, 2023, **2**, 833–854.
14 C. Zhu, C. Sun, R. Li, S. Weng, L. Fan, X. Wang, L. Chen, M. Noked and X. Fan, *ACS Energy Lett.*, 2022, **7**, 1338–1347.
15 T. Zhou, Y. Zhao, M. El Kazzi, J. W. Choi and A. Coskun, *Angew. Chem., Int. Ed.*, 2022, **61**, e202115884.
16 X. Ren, L. Zou, X. Cao, M. H. Engelhard, W. Liu, S. D. Burton, H. Lee, C. Niu, B. E. Matthews, Z. Zhu, C. Wang, B. W. Arey, J. Xiao, J. Liu, J. G. Zhang and W. Xu, *Joule*, 2019, **3**, 1662–1676.
17 S. Chen, J. Zheng, L. Yu, X. Ren, M. H. Engelhard, C. Niu, H. Lee, W. Xu, J. Xiao, J. Liu and J. G. Zhang, *Joule*, 2018, **2**, 1548–1558.
18 Q. K. Zhang, X. Q. Zhang, L. P. Hou, S. Y. Sun, Y. X. Zhan, J. L. Liang, F. S. Zhang, X. N. Feng, B. Q. Li and J. Q. Huang, *Adv. Energy Mater.*, 2022, **12**, 2200139.
19 S. Tan, Z. Shadike, J. Li, X. Wang, Y. Yang, R. Lin, A. Cresce, J. Hu, A. Hunt, I. Waluyo, L. Ma, F. Monaco, P. Cloetens, J. Xiao, Y. Liu, X. Q. Yang, K. Xu and E. Hu, *Nat. Energy*, 2022, **7**, 484–494.
20 M. Palluzzi, A. Tsurumaki, H. Adenusi, M. A. Navarra and S. Passerini, *Energy Mater.*, 2023, **3**, 300049.
21 P. Walden, *Acad. Imp. Sci. Saint-Pétersbourg*, 1914, **85**, 1800–1801.
22 J. Hwang, K. Matsumoto, C. Y. Chen and R. Hagiwara, *Energy Environ. Sci.*, 2021, **14**, 5834–5863.
23 X. Tang, S. Lv, K. Jiang, G. Zhou and X. Liu, *J. Power Sources*, 2022, **542**, 231792.
24 X. Wang, L. Jin, W. Feng, Z. Zhou and H. Zhang, *Sci. China: Chem.*, 2023, **66**, 3443–3466.
25 X. Wu, Y. Dai, N. W. Li, X. C. Chen and L. Yu, *eScience*, 2024, **4**, 100173.
26 W. Zhou, M. Zhang, X. Kong, W. Huang and Q. Zhang, *Adv. Sci.*, 2021, **8**, 2004490.
27 G. Zhang, X. Deng, J. Li, J. Wang, G. Shi, Y. Yang, J. Chang, K. Yu, S. Sen Chi, H. Wang, P. Wang, Z. Liu, Y. Gao, Z. Zheng, Y. Deng and C. Wang, *Nano Energy*, 2022, **95**, 107014.
28 R. Han, Z. Wang, D. Huang, F. Zhang, A. Pan, H. Song, Y. Wei, Y. Liu, L. Wang, Y. Li, J. Xu, J. Hu and X. Wu, *Small*, 2023, **19**, 2300571.
29 K. Matuszek, S. L. Piper, A. Brzęczek-Szafran, B. Roy, S. Saher, J. M. Pringle and D. R. MacFarlane, *Adv. Mater.*, 2024, **36**, 2313023.
30 P. Meng, Z. Yang, J. Zhang, M. Jiang, Y. Wang, X. Zhang, J. Luo and C. Fu, *Energy Storage Mater.*, 2023, **63**, 102953.
31 F. Wu, S. Fang, M. Kuenzel, A. Mullaliu, J. K. Kim, X. Gao, T. Diemant, G. T. Kim and S. Passerini, *Joule*, 2021, **5**, 2177–2194.
32 M. Y. Yang, S. V. Zybin, T. Das, B. V. Merinov, W. A. Goddard, E. K. Mok, H. J. Hah, H. E. Han, Y. C. Choi and S. H. Kim, *Adv. Energy Mater.*, 2023, **13**, 2202949.
33 H. Qi, Y. Ren, S. Guo, Y. Wang, S. Li, Y. Hu and F. Yan, *ACS Appl. Mater. Interfaces*, 2020, **12**, 591–600.
34 A. Warrington, M. Hasanpoor, A. Balkis, P. C. Howlett, O. E. Hutt, M. Forsyth and J. M. Pringle, *Energy Storage Mater.*, 2023, **63**, 102984.
35 D. A. Rakov, J. Sun, P. V. Cherepanov, K. Arano, P. C. Howlett, A. N. Simonov, F. Chen and M. Forsyth, *Energy Environ. Sci.*, 2023, **16**, 3919–3931.
36 D. A. Rakov, F. Chen, S. A. Ferdousi, H. Li, T. Pathirana, A. N. Simonov, P. C. Howlett, R. Atkin and M. Forsyth, *Nat. Mater.*, 2020, **19**, 1096–1101.
37 H. Sun, G. Zhu, Y. Zhu, M. C. Lin, H. Chen, Y. Y. Li, W. H. Hung, B. Zhou, X. Wang, Y. Bai, M. Gu, C. L. Huang, H. C. Tai, X. Xu, M. Angell, J. J. Shyue and H. Dai, *Adv. Mater.*, 2020, **32**, 2001741.
38 P. Liang, H. Sun, C. L. Huang, G. Zhu, H. C. Tai, J. Li, F. Wang, Y. Wang, C. J. Huang, S. K. Jiang, M. C. Lin, Y. Y. Li, B. J. Hwang, C. A. Wang and H. Dai, *Adv. Mater.*, 2022, **34**, 2207361.
39 Q. Liu, W. Jiang, J. Xu, Y. Xu, Z. Yang, D. J. Yoo, K. Z. Pupek, C. Wang, C. Liu, K. Xu and Z. Zhang, *Nat. Commun.*, 2023, **14**, 3678.
40 Z. Wang, Y. Sun, Y. Mao, F. Zhang, L. Zheng, D. Fu, Y. Shen, J. Hu, H. Dong, J. Xu and X. Wu, *Energy Storage Mater.*, 2020, **30**, 228–237.
41 Y. Li, F. Ding, Y. Shao, H. Wang, X. Guo, C. Liu, X. Sui, G. Sun, J. Zhou and Z. Wang, *Angew. Chem., Int. Ed.*, 2024, **63**, e202317148.
42 D. Stępień, B. Wolff, T. Diemant, G. T. Kim, F. Hausen, D. Bresser and S. Passerini, *ACS Appl. Mater. Interfaces*, 2023, **15**, 25462–25472.
43 Z. Wang, R. Han, H. Zhang, D. Huang, F. Zhang, D. Fu, Y. Liu, Y. Wei, H. Song, Y. Shen, J. Xu, J. Zheng, X. Wu and H. Li, *Adv. Funct. Mater.*, 2023, **33**, 2215065.
44 F. Ding, Y. Li, G. Zhang, H. Wang, B. Liu, C. Liu, L. Jiang, X. Sui and Z. Wang, *Adv. Mater.*, 2024, **36**, 2400177.

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45 U. Pal, D. Rakov, B. Lu, B. Sayahpour, F. Chen, B. Roy, D. R. MacFarlane, M. Armand, P. C. Howlett, Y. S. Meng and M. Forsyth, *Energy Environ. Sci.*, 2022, **15**, 1907–1919.
46 K. Ding, E. J. Begin, S. Yuan, M. Zhong, Y. Wang, Y. Zhang, X. Zeng, J. L. Bao and Y. Wang, *Adv. Energy Mater.*, 2023, **13**, 2302443.
47 S. Chen, J. J. Fan, Z. Cui, L. Tan, D. Ruan, X. Zhao, J. Jiang, S. Jiao and X. Ren, *Angew. Chem., Int. Ed.*, 2023, **62**, e202219310.
48 Y. Zou, Z. Ma, G. Liu, Q. Li, D. Yin, X. Shi, Z. Cao, Z. Tian, H. Kim, Y. Guo, C. Sun, L. Cavallo, L. Wang, H. N. Alshareef, Y. K. Sun and J. Ming, *Angew. Chem., Int. Ed.*, 2023, **62**, e202216189.
49 X. Liu, A. Mariani, H. Adenusi and S. Passerini, *Angew. Chem., Int. Ed.*, 2023, **62**, e202219318.
50 H. Jia, J. M. Kim, P. Gao, Y. Xu, M. H. Engelhard, B. E. Matthews, C. Wang and W. Xu, *Angew. Chem., Int. Ed.*, 2023, **62**, e202218005.
51 Z. Wang, F. Zhang, Y. Sun, L. Zheng, Y. Shen, D. Fu, W. Li, A. Pan, L. Wang, J. Xu, J. Hu and X. Wu, *Adv. Energy Mater.*, 2021, **11**, 2003752.
52 X. Liu, T. Diemant, A. Mariani, X. Dong, M. E. Di Pietro, A. Mele and S. Passerini, *Adv. Mater.*, 2022, **34**, 2207155.
53 X. Liu, A. Mariani, T. Diemant, M. E. Di Pietro, X. Dong, A. Mele and S. Passerini, *Adv. Mater.*, 2024, **36**, 2309062.
54 A. Sun, H. Tu, Z. Sun, Z. He, Y. Wang, J. Wang, Y. Zheng, F. Zhu, L. Wang, F. Mushtaq, P. Xue, J. Liu and M. Liu, *ACS Energy Lett.*, 2024, **9**, 2545–2553.
55 Y. Cai, Q. Zhang, Y. Lu, Z. Hao, Y. Ni and J. Chen, *Angew. Chem., Int. Ed.*, 2021, **60**, 25973–25980.
56 S. Lee, K. Park, B. Koo, C. Park, M. Jang, H. Lee and H. Lee, *Adv. Funct. Mater.*, 2020, **30**, 2003132.
57 Z. Wang, H. Zhang, J. Xu, A. Pan, F. Zhang, L. Wang, R. Han, J. Hu, M. Liu and X. Wu, *Adv. Funct. Mater.*, 2022, **32**, 2112598.
58 X. Liu, A. Mariani, M. Zarrabeitia, M. Enrica, D. Pietro, X. Dong, G. Antonio, A. Mele and S. Passerini, *Energy Storage Mater.*, 2022, **44**, 370–378.
59 H. Tu, L. Li, Z. Wang, J. Wang, H. Lin, M. Wang, C. Yan and M. Liu, *ACS Nano*, 2022, **16**, 16898–16908.
60 X. Liu, A. Mariani, T. Diemant, M. E. Di Pietro, X. Dong, M. Kuenzel, A. Mele and S. Passerini, *Adv. Energy Mater.*, 2022, **12**, 2200862.
61 X. Liu, A. Mariani, T. Diemant, X. Dong, P. H. Su and S. Passerini, *Angew. Chem., Int. Ed.*, 2023, **62**, e202305840.
62 X. Liu, A. Mariani, T. Diemant, M. E. Di Pietro, X. Dong, P. H. Su, A. Mele and S. Passerini, *ACS Energy Lett.*, 2024, **9**, 3049–3057.
63 W. Zou, J. Zhang, M. Liu, J. Li, Z. Ren, W. Zhao, Y. Zhang, Y. Shen and Y. Tang, *Adv. Mater.*, 2024, **36**, 2400537.
64 Y. Cai, Y. Hou, Y. Lu, Q. Zhang, Z. Yan and J. Chen, *Angew. Chem., Int. Ed.*, 2023, **62**, e202218014.
65 Y. Zheng, H. Ji, T. Qian, S. Li, J. Liu, J. Zhou, Z. Wang, Y. Li and C. Yan, *Nano Lett.*, 2023, **23**, 3181–3188.
66 X. Liu, M. Zarrabeitia, A. Mariani, X. Gao, H. M. Schütz, S. Fang, T. Bizien, G. A. Elia and S. Passerini, *Small Methods*, 2021, **5**, 2100168.
67 Z. Wang, H. Zhang, R. Han, J. Xu, A. Pan, F. Zhang, D. Huang, Y. Wei, L. Wang, H. Song, Y. Liu, Y. Shen, J. Hu and X. Wu, *ACS Sustain. Chem. Eng.*, 2022, **10**, 12023–12029.
68 Y. Liu, S. Lu, Z. Wang, J. Xu, S. Weng, J. Xue, H. Tu, F. Zhang, L. Liu, Y. Gao, H. Li, J. Zheng and X. Wu, *Adv. Funct. Mater.*, 2024, **34**, 2312295.
69 X. Hu, E. Matios, Y. Zhang, C. Wang, J. Luo and W. Li, *Angew. Chem., Int. Ed.*, 2021, **133**, 6043–6048.
70 Y. Huang, B. Cao, Z. Geng and H. Li, *Acc. Mater. Res.*, 2024, **5**, 184–193.
71 H. Yang, J. Li, Z. Sun, R. Fang, D. Wang, K. He, H. Cheng and F. Li, *Energy Storage Mater.*, 2020, **30**, 113–129.
72 Z. Li, S. Wang, J. Shi, Y. Liu, S. Zheng, H. Zou, Y. Chen, W. Kuang, K. Ding, L. Chen, Y. Lan, Y. Cai and Q. Zheng, *Energy Storage Mater.*, 2022, **47**, 262–270.
73 J. Wan, J. Xie, X. Kong, Z. Liu, K. Liu, F. Shi, A. Pei, H. Chen, W. Chen, J. Chen, X. Zhang, L. Zong, J. Wang, L.-Q. Chen, J. Qin and Y. Cui, *Nat. Nanotechnol.*, 2019, **14**, 705–711.
74 M. Martinez-Ibañez, N. Boaretto, A. Santiago, L. Meabe, X. Wang, O. Zugazua, I. Raposo, M. Forsyth, M. Armand and H. Zhang, *J. Power Sources*, 2023, **557**, 232554.
75 X. Ma, J. Yu, Y. Hu, J. Texter and F. Yan, *Ind. Chem. Mater.*, 2023, **1**, 39–59.
76 B. Tong, Z. Song, W. Feng, J. Zhu, H. Yu, X. Huang, M. Armand, Z. Zhou and H. Zhang, *Adv. Energy Mater.*, 2023, **13**, 2204085.
77 C. Heubner, S. Maletti, H. Auer, J. Hüttl, K. Voigt, O. Lohrberg, K. Nikolowski, M. Partsch and A. Michaelis, *Adv. Funct. Mater.*, 2021, **31**, 2106608.
78 N. W. Utomo, S. Hong, R. Sinha, K. Kim, Y. Deng, P. Ochonma, M. G. Kitahata, R. Garcia-Mendez, Y. L. Joo and L. A. Archer, *Sci. Adv.*, 2024, **10**, eado4719.
79 G. Eshetu, D. Mecerreyes, M. Forsyth, H. Zhang and M. Armand, *Mol. Syst. Des. Eng.*, 2019, **4**, 294–309.
80 J. Ma, X. Ma, H. Zhang, F. Chen, X. Guan, J. Niu and X. Hu, *J. Membr. Sci.*, 2022, **659**, 120811.
81 L. Gao, W. Jiang, X. Zhang, Y. Sun, K. Chen, W. Li, H. Xie and J. Liu, *Chem. Eng. J.*, 2024, **479**, 147822.
82 T. Huang, M. Long, X. Wang, G. Wu and Y. Wang, *Chem. Eng. J.*, 2019, **375**, 122062.
83 A. Shaplov, R. Marcilla and D. Mecerreyes, *Electrochim. Acta*, 2015, **175**, 18–34.
84 H. Chang, W. Li, H. Liu, H. Hu, W. Liu, Y. Jin and G. Cui, *Chem. Eng. J.*, 2024, **481**, 148602.
85 D. Zhou, R. Liu, J. Zhang, X. Qi, Y. He, B. Li, Q. Yang, Y. Hu and F. Kang, *Nano Energy*, 2017, **33**, 45–54.
86 L. Liang, X. Chen, W. Yuan, H. Chen, H. Liao and Y. Zhang, *ACS Appl. Mater. Interfaces*, 2021, **13**, 25410–25420.
87 V. Delhorbe, D. Bresser, H. Mendil-Jakani, P. Rannou, L. Bernard, T. Gutel, S. Lyonnard and L. Picard, *Macromolecules*, 2017, **50**, 4309–4321.
88 F. Makhlooghiazad, L. Miguel Guerrero Mejía, G. Rollo-Walker, D. Kourati, M. Galceran, F. Chen, M. Deschamps, P. Howlett, L. A. O'Dell and M. Forsyth, *J. Am. Chem. Soc.*, 2024, **146**, 1992–2004.

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89 X. Wang, F. Chen, G. M. A. Girard, H. Zhu, D. R. MacFarlane, D. Mecerreyes, M. Armand, P. C. Howlett and M. Forsyth, *Joule*, 2019, **3**, 2687–2702.
90 X. Lin, S. Xu, Y. Tong, X. Liu, Z. Liu, P. Li, R. Liu, X. Feng, L. Shi and Y. Ma, *Mater. Horiz.*, 2023, **10**, 859–868.
91 F. Zhang, Y. Sun, Z. Wang, D. Fu, J. Li, J. Hu, J. Xu and X. Wu, *ACS Appl. Mater. Interfaces*, 2020, **12**, 23774–23780.
92 L. Liu, J. Xue, Y. Liu, S. Lu, S. Weng, Z. Wang, F. Zhang, D. Fu, J. Xu and X. Wu, *ACS Appl. Mater. Interfaces*, 2024, **16**, 8895–8902.
93 N. Kiriy, S. Özenler, P. Voigt, O. Kobsch, J. Meier-Haack, K. Arnhold, A. Janke, U. L. Muza, M. Geisler and A. Lederer, *Int. J. Mol. Sci.*, 2024, **25**, 1595.
94 Y. Li, Z. Sun, L. Shi, S. Lu, Z. Sun, Y. Shi, H. Wu, Y. Zhang and S. Ding, *Chem. Eng. J.*, 2019, **375**, 121925.
95 X. Chen, L. Liang, W. Hu, H. Liao and Y. Zhang, *J. Power Sources*, 2022, **542**, 231766.
96 J. Liu, Y. Xu, F. Xu, J. Li, Y. Chen, J. Qiao, Y. Han, Y. Ren and B. Lin, *Ionics*, 2023, **29**, 2249–2259.
97 Y. Cui, G. Yu, R. Liu, D. Miao and D. Wu, *Chin. J. Chem.*, 2023, **41**, 2848–2854.
98 X. Wang, G. M. Girard, H. Zhu, R. Yunis, D. R. MacFarlane, D. Mecerreyes, A. J. Bhattacharyya, P. C. Howlett and M. Forsyth, *ACS Appl. Energy Mater.*, 2019, **2**, 6237–6245.
99 M. Yao, Q. Ruan, Y. Wang, L. Du, Q. Li, L. Xu, R. Wang and H. Zhang, *Adv. Funct. Mater.*, 2023, **33**, 2213702.
100 R. Zhao, J. Yang, B. Wang, Z. Ma, L. Pan and Y. Li, *Chin. J. Chem.*, 2023, **41**, 2493–2501.
101 Q. Liu, L. Li, G. Liu, X. He, Y. Niu and G. Li, *J. Power Sources*, 2024, **592**, 233897.
102 A. J. D'Angelo and M. J. Panzer, *Adv. Energy Mater.*, 2018, **8**, 1801646.
103 J. H. Shin, W. A. Henderson and S. Passerini, *J. Electrochem. Soc.*, 2005, **152**, A978.
104 M. Zhu, J. Wu, Y. Wang, M. Song, L. Long, S. H. Siyal, X. Yang and G. Sui, *J. Energy Chem.*, 2019, **37**, 126–142.
105 J. Qian, B. Jin, Y. Li, X. Zhan, Y. Hou and Q. Zhang, *J. Energy Chem.*, 2021, **56**, 420–437.
106 L. Anh and D. Kim, *ACS Appl. Energy Mater.*, 2019, **2**, 2585–2595.
107 Y. Yuan, X. Peng, B. Wang, K. Xue, Z. Li, Y. Ma, B. Zheng, Y. Song and H. Lu, *J. Mater. Chem. A*, 2023, **11**, 1301–1311.
108 H. Wang, X. Li, Q. Zeng, Z. Li, Y. Liu, J. Guan, Y. Jiang, L. Chen, Y. Cao and R. Li, *Energy Storage Mater.*, 2024, **66**, 103188.
109 T. Chen, W. Kong, Z. Zhang, L. Wang, Y. Hu, G. Zhu, R. Chen, L. Ma, W. Yan, Y. Wang, J. Liu and Z. Jin, *Nano Energy*, 2018, **54**, 17–25.
110 Z. Wang, Y. Wang, P. Zhai, P. Poldorn, S. Jungsuttiwong and S. Yuan, *J. Energy Chem.*, 2022, **75**, 340–348.
111 D. Xu, D. Zhao, X. Niu, T. Wang and Z. Yang, *Chem. Eng. J.*, 2024, **490**, 151780.
112 Q. Yang, Z. Zhang, X. Sun, Y. Hu, H. Xing and S. Dai, *Chem. Soc. Rev.*, 2018, **47**, 2020–2064.
113 J. Weston and B. Steele, *Solid State Ionics*, 1982, **7**, 75–79.
114 M. Gandolfo, D. Versaci, C. Francia, S. Bodoardo and J. Amici, *Electrochim. Acta*, 2023, **463**, 142857.
115 D. Zhang, X. Xu, X. Huang, Z. Shi, Z. Wang, Z. Liu, R. Hu, J. Liu and M. Zhu, *J. Mater. Chem. A*, 2020, **8**, 18043–18054.
116 D. Yu, Z. Ma, Z. Liu, X. Jiang, H. A. Younus, X. Wang and S. Zhang, *Chem. Eng. J.*, 2023, **457**, 141043.
117 T. Kim, S. Yun, J. Chae, H. Kim and U. Choi, *ACS Appl. Energy Mater.*, 2023, **7**, 48–60.
118 X. Pan, Q. Hou, L. Liu, J. Zhang, M. An and P. Yang, *Ionics*, 2021, **27**, 2045–2051.
119 Z. Shen, Y. Cheng, S. Sun, X. Ke, L. Liu and Z. Shi, *Carbon Energy*, 2021, **3**, 482–508.
120 M. Liu, S. Zhang, E. van Eck, C. Wang, S. Ganapathy and M. Wagemaker, *Nat. Nanotechnol.*, 2022, **17**, 959–967.
121 L. Zhu, J. Chen, Y. Wang, W. Feng, Y. Zhu, S. Lambregts, Y. Wu, C. Yang, E. van Eck, L. Peng, A. Kentgens, W. Tang and Y. Xia, *J. Am. Chem. Soc.*, 2024, **146**, 6591–6603.
122 W. Lin, H. Yuan, C. Tian, M. Song, T. Huang and A. Yu, *Energy Storage Mater.*, 2024, **70**, 103472.
123 S. Xue, S. Chen, Y. Fu, H. Zhu, Y. Ji, Y. Song, F. Pan and L. Yang, *Small*, 2023, **19**, 2305326.
124 R. Xu, X. Shen, X. X. Ma, C. Yan, X. Q. Zhang, X. Chen, J. F. Ding and J. Q. Huang, *Angew. Chem., Int. Ed.*, 2021, **60**, 4215–4220.
125 K. Xu, *J. Power Sources*, 2023, **559**, 232652.
126 S. Liu, X. Ji, N. Piao, J. Chen, N. Eidson, J. Xu, P. Wang, L. Chen, J. Zhang, T. Deng, S. Hou, T. Jin, H. Wan, J. Li, J. Tu and C. Wang, *Angew. Chem., Int. Ed.*, 2021, **60**, 3661–3671.
127 N. Piao, S. Liu, B. Zhang, X. Ji, X. Fan, L. Wang, P. F. Wang, T. Jin, S. C. Liou, H. Yang, J. Jiang, K. Xu, M. A. Schroeder, X. He and C. Wang, *ACS Energy Lett.*, 2021, **6**, 1839–1848.
128 X. Ma, J. Yu, Y. Hu, J. Texter and F. Yan, *Ind. Chem. Mater.*, 2023, **4**, 101379.
129 G. Huang, Y. Liao, X. Zhao, X. Jin, Z. Zhu, M. Guan and Y. Li, *Adv. Funct. Mater.*, 2022, **33**, 2211364.
130 W. Fang, Z. Wen, L. Chen, Z. Qin, J. Li, Z. Zheng, Z. Weng, G. Wu, N. Zhang, X. Liu, X. Yuan and G. Chen, *Nano Energy*, 2022, **104**, 107881.
131 Y. Gao, G. Wu, W. Fang, Z. Qin, T. Zhang, J. Yan, Y. Zhong, N. Zhang and G. Chen, *Angew. Chem., Int. Ed.*, 2024, **63**, e202403668.
132 J. Wang, J. Yang, Q. Xiao, J. Zhang, T. Li, L. Jia, Z. Wang, S. Cheng, L. Li, M. Liu, H. Liu, H. Lin and Y. Zhang, *Adv. Funct. Mater.*, 2021, **31**, 2007434.
133 F. Zhang, Z. Wang, L. Wang, W. Li, A. Pan, H. Song, J. Xu, J. Hu and X. Wu, *Chem. Eng. J.*, 2022, **435**, 135101.
134 J. Yu, X. Ma, X. Zou, Y. Hu, M. Yang, J. Yang, S. Sun and F. Yan, *Energy Environ. Sci.*, 2024, **17**, 4519–4530.
135 Y. Qin, H. Wang, J. Zhou, R. Li, C. Jiang, Y. Wan, X. Wang, Z. Chen, X. Wang, Y. Liu, B. Guo and D. Wang, *Angew. Chem., Int. Ed.*, 2024, **63**, e202402456.
136 J. S. Kim, G. Yoon, S. Kim, S. Sugata, N. Yashiro, S. Suzuki, M. J. Lee, R. Kim, M. Badding, Z. Song, J. M. Chang and D. Im, *Nat. Commun.*, 2023, **14**, 782.
137 M. Shen, L. Zhang, C. Li, X. Feng, R. Zheng, H. Sun, Z. Wang and Y. Liu, *J. Electroanal. Chem.*, 2024, **957**, 118126.
138 J. Wang, L. Li, H. Hu, H. Hu, Q. Guan, M. Huang, L. Jia, H. Adenusi, K. V. Tian and J. Zhang, *ACS Nano*, 2022, **16**, 17729–17760.