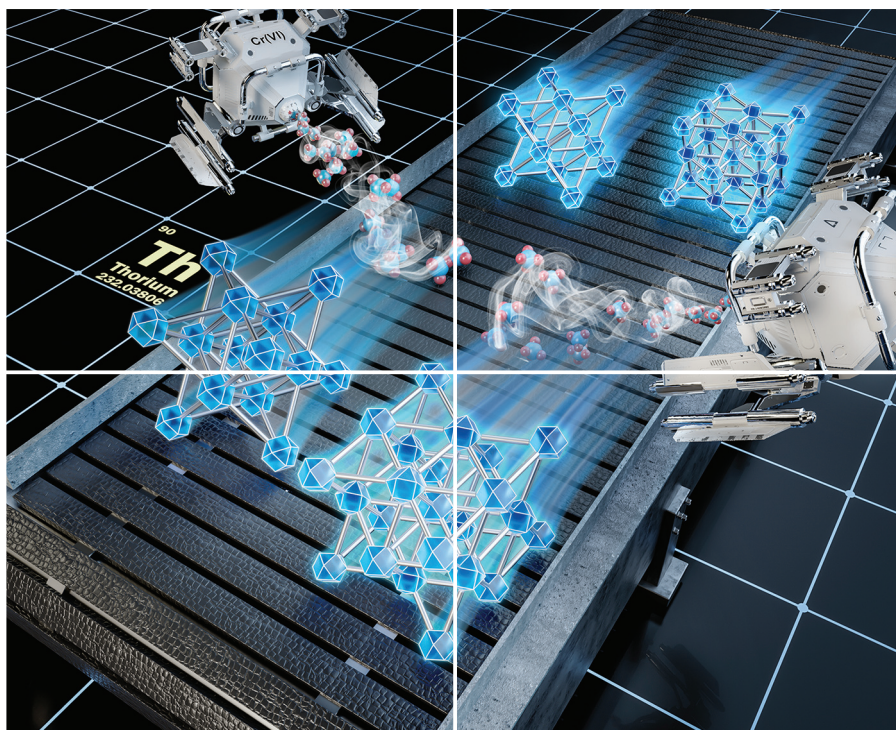


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# Topological control of metal–organic frameworks toward highly sensitive and selective detection of chromate and dichromate†

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Luminescent metal–organic frameworks (LMOFs) have been extensively used as sensitive and selective sensors for carcinogenic chromium(vi) oxyanions. However, the correlation between the MOF structure and the Cr(vi) sensing efficacy remains underexplored, hindering the rational design of MOFs for Cr(vi) sensing. Herein, we overcome this challenge by judiciously selecting the rarely touched thorium cations and a tetraphenylethene-based ligand for building two novel thorium-based MOFs, **Th-BCTPE-1** and **Th-BCTPE-2**. Despite being built from identical precursors, **Th-BCTPE-1** shows more sensitive luminescence quenching responses to  $\text{CrO}_4^{2-}$  and  $\text{Cr}_2\text{O}_7^{2-}$  than **Th-BCTPE-2**, producing the second highest quenching constants of  $\text{CrO}_4^{2-}$  among all LMOF-based Cr(vi) sensors. We further decoupled the influences of different structural variables and the corresponding physiochemical properties, including porosity, BET surface area, photoluminescence quantum yield, and adsorption capacity, showing that the sensing efficacy of Cr(vi) oxyanions is more relevant to the adsorption capacity and the degree of the inner filter effect than the rest of the variables in this system.

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## Introduction

Chromates and dichromates have been extensively used to improve the corrosion resistance of active metals *via* the formation of stable chromate conversion coating.<sup>1</sup> Moreover, Cr(vi) oxyanions are essential in advanced nuclear cycles since chromate and dichromate have been used as corrosion inhibitors in cooling water for nuclear reactors.<sup>2,3</sup> Cooling water with an approximate Cr(vi) concentration of  $700 \mu\text{g kg}^{-1}$  is discharged from the reactors to natural water bodies after cooling in retention basins.<sup>4</sup> As a consequence, high-level Cr(vi) contamination (ranging up to nearly  $300 \text{mg kg}^{-1}$ ) has been identified in the groundwater at the Hanford site and the nearby

Columbia river, which imposes a potential hazard to human health due to its high carcinogenicity.<sup>4,5</sup> Furthermore, the presence of chromate in high-level waste (HLW) leads to the formation of undesirable spinel crystals during the vitrification of nuclear waste, which increases the anticipated cost and risk of HLW vitrification.<sup>6</sup> Hence, facile detection and the subsequent effective sequestration of chromate and dichromate from contaminated water and HLW are highly indispensable.<sup>7–11</sup> The concentration of Cr(vi) solution can be determined colorimetrically by the diphenylcarbazide method.<sup>4</sup> However, this technology suffers from low sensitivity (limit of detection of  $0.5 \text{mg kg}^{-1}$ ). Instrumental techniques including inductively coupled plasma atomic emission spectroscopy (ICP-AES) and inductively coupled plasma mass spectrometry (ICP-MS) are advantageous in terms of limit of detection (LOD)<sup>12</sup> but they are unable to distinguish the toxic Cr(vi) from the benign Cr(III) ion, the latter of which is the dominant natural background species of chromium in sediments.

As a class of porous inorganic–organic hybrid materials, metal–organic frameworks (MOFs) are scientifically compelling and functionally evolving with considerable advances made in various fields including separation, sensing, and catalysis.<sup>13–18</sup> Incorporating luminescent metal nodes and/or organic linkers within MOFs engenders the sensing capacities of MOFs in a turn-off, turn-on, or fluorochromic manner, making them suitable for chromate/dichromate sensing.<sup>19–22</sup>

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Furthermore, the highly tunable compositions, structures, and pore functionalities of MOFs give rise to a series of chromate/dichromate sensors with high sensitivity and excellent selectivity.<sup>23–28</sup> Although significant progress in the development of MOFs for Cr(vi) detection has been achieved, understanding the relationship between the MOF structure and the sensing efficacy remains in its infancy, which hinders the rational design of MOFs for Cr(vi) sensing. The study by Sun and coworkers indicated that modifying the pore shape and volume of MOFs *via* altering the linker shape have a notable influence on the detection sensitivity of dichromate.<sup>25</sup> Moreover, a mixed-ligand strategy gave rise to three cadmium coordination polymers, of which the one-dimensional polymer exhibited the best sensing performance.<sup>29</sup> However, both the organic linkers and the topologies of MOFs are changed in these systems, rendering the decoupling of the influence of each variable on the luminescence properties rather difficult. Therefore, an outstanding challenge in obtaining the structure–property correlation is to alter the underlying topologies of MOFs without introducing additional variables.

We have recently undertaken a study on the control of the polymorphism of MOFs by utilizing thorium-based metal–organic frameworks (Th-MOFs) as a platform.<sup>30</sup> One of the key merits of Th-MOFs is that the appropriate “hardness” of Th<sup>4+</sup> can afford different secondary building units (SBUs) with a wide range of nuclearity and correspondingly diverse topologies.<sup>31–35</sup> Besides, the Th<sup>4+</sup> cation typically adopts a nine-coordinate capped square antiprism geometry, which provides an additional coordinating site on its capping addenda for creating new structures.<sup>36–39</sup> This can be exemplified by a series of Th-MOFs that feature five unique topologies even though they are built from the same 2'-methyl-[1,1':4',1''-terphenyl]-4,4''-dicarboxylate (MeTPDC<sup>2-</sup>) ligand.<sup>38</sup> Such topological polymorphism of Th-MOFs facilitates a better understanding of how the structure of MOF, in isolation, influences the sensing efficacy of Cr(vi) oxyanions.

Herein, we showcase a synthetic modulation approach to obtain two topologically distinct Th-MOFs, [Th<sub>6</sub>(OH)<sub>4</sub>(O)<sub>4</sub>(H<sub>2</sub>O)<sub>6</sub>](BCTPE)<sub>6</sub>(DMF)<sub>18</sub>(H<sub>2</sub>O)<sub>9</sub> (**Th-BCTPE-1**) and [Th<sub>6</sub>(OH)<sub>4</sub>(O)<sub>4</sub>(H<sub>2</sub>O)<sub>6</sub>](BCTPE)<sub>5</sub>(HCOO)<sub>2</sub>(DMF)<sub>26</sub>(H<sub>2</sub>O)<sub>32</sub> (**Th-BCTPE-2**), built from the same (*E*)-4,4'-(1,2-diphenylethene-1,2-diyl)dibenzoate linker (BCTPE<sup>2-</sup>). As a  $\pi$ -electron-rich dicarboxylate ligand, H<sub>2</sub>BCTPE was used as a fluorophore to engender the luminometric sensing capacities of **Th-BCTPE-1** and **Th-BCTPE-2** for chromate and dichromate. Notably, the UiO-type MOF **Th-BCTPE-1** features greater quenching constants ( $K_{SV}$ ) and a lower limit of detection (LOD) than **Th-BCTPE-2** for both chromate and dichromate. The polymorphism of these Th-MOFs and the corresponding different sensing efficacies show the profound effect of structure on Cr(vi) sensing and shed light on the future rational design of highly efficient Cr(vi) sensors.

## Results and discussion

By modifying the stoichiometry, modulator, and reaction temperature, solvothermal reactions between the identical pre-

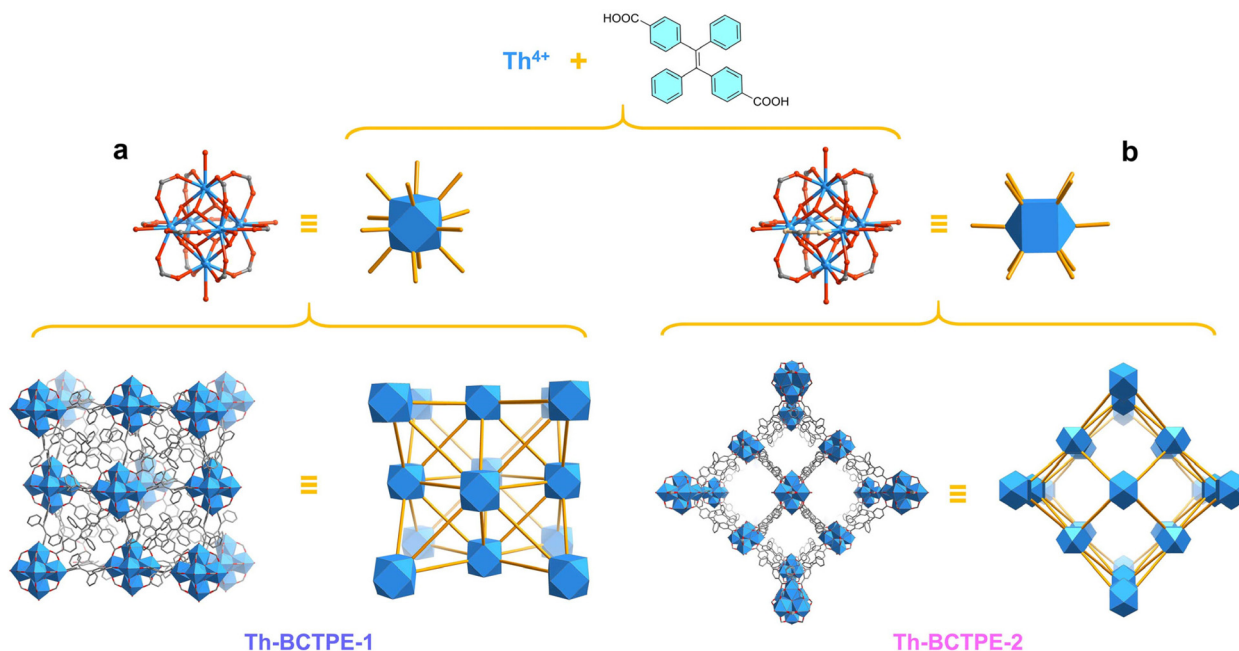
cursors of Th(NO<sub>3</sub>)<sub>4</sub>·6H<sub>2</sub>O and (*E*)-4,4'-(1,2-diphenylethene-1,2-diyl)dibenzoic acid (H<sub>2</sub>BCTPE) allow for the isolation of two different Th-MOFs, **Th-BCTPE-1** and **Th-BCTPE-2**. When trifluoroacetic acid was selected as a modulator, a reaction at 80 °C gave rise to octahedral single-crystals of **Th-BCTPE-1** (Fig. S1a†). Incorporating nitric acid in the synthesis at 120 °C resulted in the crystallization of prismatic crystals of **Th-BCTPE-2** (Fig. S1b†).

Single-crystal X-ray diffraction (SCXRD) study revealed that **Th-BCTPE-1** adopts a typical UiO type structure with the cubic *Fm* $\bar{3}$ *m* space group and *fcu* topology (Table S1†). The asymmetric unit of its framework is composed of one crystallographically unique 1/8 Th<sup>4+</sup>, 1/6  $\mu_3$ -OH/ $\mu_3$ -O, 1/8 coordinating H<sub>2</sub>O (1/48 [Th<sub>6</sub>( $\mu_3$ -OH)<sub>4</sub>( $\mu_3$ -O)<sub>4</sub>(H<sub>2</sub>O)<sub>6</sub>]<sup>12+</sup>), and 1/8 BCTPE<sup>2-</sup>. As shown in Fig. 1a, the secondary building unit (SBU) of **Th-BCTPE-1** is built from six Th cations interconnected by four  $\mu_3$ -OH and four  $\mu_3$ -O groups and further decorated by six H<sub>2</sub>O molecules. The afforded [Th<sub>6</sub>( $\mu_3$ -OH)<sub>4</sub>( $\mu_3$ -O)<sub>4</sub>(H<sub>2</sub>O)<sub>6</sub>]<sup>12+</sup> cluster is bridged by twelve BCTPE<sup>2-</sup> dicarboxylate linkers with the neighbouring clusters, generating a 12-connected SBU. Two types of voids, namely an octahedral (9.0 Å) and a tetrahedral (6.2 Å) void, were identified. These cavities are occupied by disordered DMF and H<sub>2</sub>O guest species, accounting for a total volume of 34.1% as estimated by PLATON.<sup>40</sup>

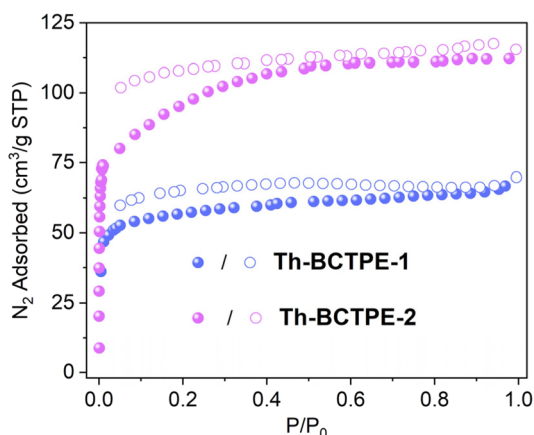
**Th-BCTPE-2** crystallizes in the *P*<sub>4</sub><sub>2</sub>/*mmc* space group and its asymmetric unit is assembled from one 1/8 Th<sub>6</sub>( $\mu_3$ -OH)<sub>4</sub>( $\mu_3$ -O)<sub>4</sub>(H<sub>2</sub>O)<sub>6</sub>(HCOO)<sub>2</sub> cluster, one 1/2 and one 1/8 BCTPE<sup>2-</sup> anion. In sharp contrast to **Th-BCTPE-1**, the SBU of **Th-BCTPE-2** is assembled from a [Th<sub>6</sub>( $\mu_3$ -OH)<sub>4</sub>( $\mu_3$ -O)<sub>4</sub>(H<sub>2</sub>O)<sub>6</sub>]<sup>12+</sup> core chelated with ten BCTPE<sup>2-</sup> ligands and two HCOO<sup>-</sup> anions (Fig. 1b). It is worth noting that HCOO<sup>-</sup> anions originated from the hydrolyzation of DMF molecules in the presence of nitric acid with extended reaction time. Unlike the bitopic BCTPE<sup>2-</sup> ligands that interconnect the neighbouring Th<sub>6</sub> SBUs, the HCOO<sup>-</sup> groups in **Th-BCTPE-2** terminate the bridging of SBUs. Thus, each Th<sub>6</sub> SBU is connected to ten neighbouring ones, giving a final three-dimensional porous network with the *bcu* topology. Such a structure renders open channels of approximately 8.1 Å × 8.1 Å, that result in a much higher solvent-accessible volume of 64.7% than that of **Th-BCTPE-1**.

**Th-BCTPE-1** and **Th-BCTPE-2** can be obtained as pure phases, as confirmed by the similar powder X-ray diffraction (PXRD) patterns of the as-synthesized samples and the simulated ones (Fig. S2†). The permanent porosities of these Th-MOFs were examined by N<sub>2</sub> adsorption/desorption isotherms measured at 77 K. Both **Th-BCTPE-1** and **Th-BCTPE-2** exhibited typical type I isotherms with saturated uptakes of 69.8 and 115.4 cm<sup>3</sup> g<sup>-1</sup>, respectively, corresponding to Brunauer–Emmett–Teller (BET) surface areas of 188 and 316 m<sup>2</sup> g<sup>-1</sup> (Fig. 2). This trend matches well with the tendency of void volumes obtained from PLATON. The desorption branches of both materials feature slight hysteresis loops, which are characteristic of moderate interactions between N<sub>2</sub> and the frameworks. Thermogravimetric analysis (TGA) study revealed that **Th-BCTPE-1** and **Th-BCTPE-2** are stable up to 490 and





**Fig. 1** (a) The 12-connected  $\text{Th}_6(\mu_3\text{-OH})_4(\mu_3\text{-O})_4(\text{H}_2\text{O})_6(\text{R-COO})_{12}$  SBU, 3D structure, and simplified topology of **Th-BCTPE-1**. (b) The 10-connected  $\text{Th}_6(\mu_3\text{-OH})_4(\mu_3\text{-O})_4(\text{H}_2\text{O})_6(\text{HCOO})_2(\text{R-COO})_{10}$  SBU, 3D structure, and simplified topology of **Th-BCTPE-2**. Colour code: Th atoms are shown in blue, O atoms are in red, C atoms from the BCTPE<sup>2-</sup> ligands are in grey, and O atoms from the HCOO<sup>-</sup> ligand are in light yellow.



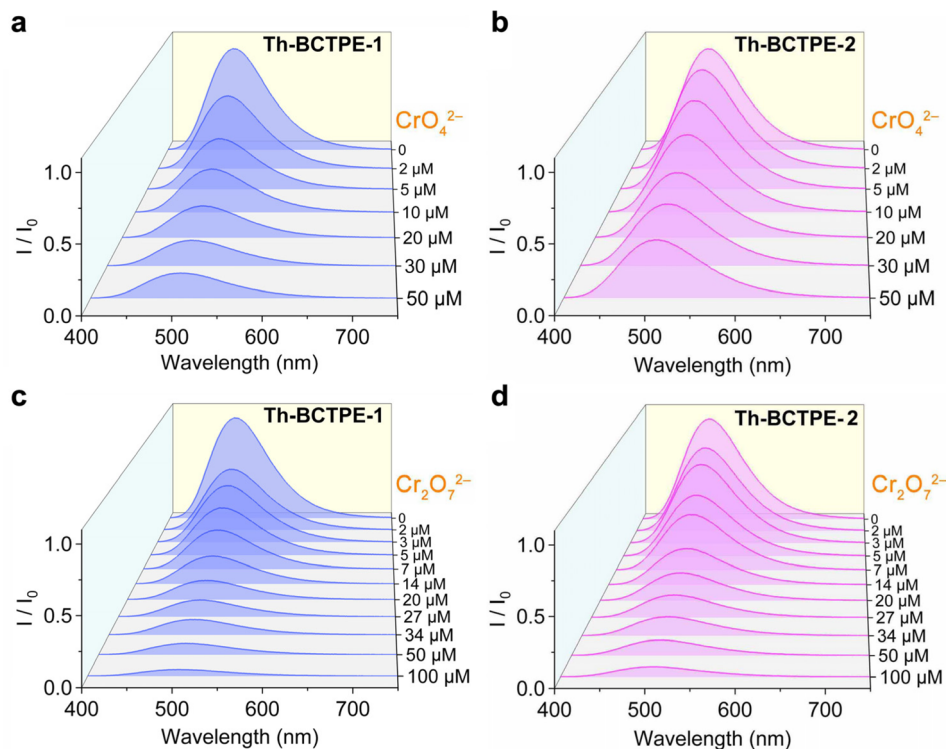
**Fig. 2**  $\text{N}_2$  adsorption/desorption isotherms of **Th-BCTPE-1** and **Th-BCTPE-2** at 77 K.

510 °C, respectively (Fig. S3†). Both Th-MOFs undergo three stages of weight loss, and the first and second stages correspond to the departure of guest species including DMF and  $\text{H}_2\text{O}$ . **Th-BCTPE-2** features a greater amount of solvent loss ( $w\% = 39.9\%$ ) due to its larger void space than that of **Th-BCTPE-1**.

Despite the tremendous effort focused on the aggregation-induced emission (AIE) of tetraphenylethene, there has been only one MOF report to date integrating the BCTPE<sup>2-</sup> ligand, a dicarboxylate derivative of TPE.<sup>41</sup> The luminescence properties of **Th-BCTPE-1** and **Th-BCTPE-2** were initially investigated by collecting their emission spectra. Upon 365 nm UV excitation,

**Th-BCTPE-1** and **Th-BCTPE-2** exhibit green luminescence with emission bands centered at 495 and 500 nm, respectively, which are red-shifted in comparison with the blue emission ( $\lambda_{\text{max}} = 467$  nm) of the pure  $\text{H}_2\text{BCTPE}$  ligand (Fig. S4†). The luminescence of these Th-MOFs can be ascribed to the matrix coordination-induced emission (MCIE), whereas that of  $\text{H}_2\text{BCTPE}$  could result from the AIE effect.<sup>42</sup> Moreover, the photoluminescence quantum yields (PLQYs) under ambient conditions were measured to be 50.53% and 53.07% for **Th-BCTPE-1** and **Th-BCTPE-2**, respectively, both of which are higher than 39.19% for  $\text{H}_2\text{BCTPE}$ . This observation indicates that the ordered and spatial anchoring of fluorescent BCTPE<sup>2-</sup> moieties inside rigid Th-MOFs can efficiently eliminate possible dissipative processes and turn on fluorescence.<sup>43–45</sup> In addition, **Th-BCTPE-1**, **Th-BCTPE-2** and  $\text{H}_2\text{BCTPE}$  show bi-exponential luminescence decays with average lifetimes of 3.21, 2.45, and 2.87 ns, respectively (Fig. S5†).

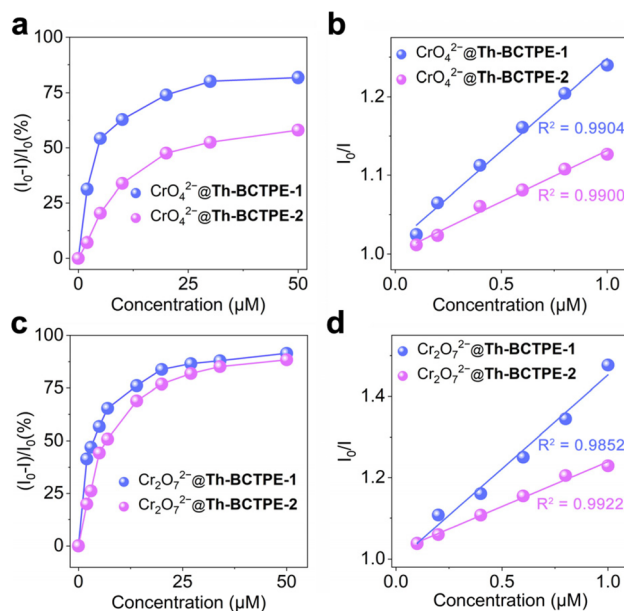
Given the intense emissions and high PLQYs of both materials, we sought to investigate the potential utility of **Th-BCTPE-1** and **Th-BCTPE-2** for chromate and dichromate sensing. Moreover, their variety and multiplicity in terms of structures, void spaces, and surface areas make them ideal candidates for elucidating the correlations of the MOF structure with the Cr(vi) sensing efficacy. In this regard, equivalent amounts (0.16  $\mu\text{mol}$ ) of **Th-BCTPE-1** or **Th-BCTPE-2** were dispersed in 2 mL  $\text{K}_2\text{CrO}_4$  or  $\text{K}_2\text{Cr}_2\text{O}_7$  solution with increasing concentrations. Both **Th-BCTPE-1** and **Th-BCTPE-2** were finely ground to make the materials homogeneously dispersed in Cr(vi) solutions and to eliminate the effect of morphology difference. As shown in Fig. 3, the emissions of both Th-MOFs



**Fig. 3** Normalized luminescence spectra of (a) **Th-BCTPE-1** as a function of increasing  $\text{CrO}_4^{2-}$  concentration, (b) **Th-BCTPE-2** as a function of increasing  $\text{CrO}_4^{2-}$  concentration, (c) **Th-BCTPE-1** as a function of increasing  $\text{Cr}_2\text{O}_7^{2-}$  concentration, and (d) **Th-BCTPE-2** as a function of increasing  $\text{Cr}_2\text{O}_7^{2-}$  concentration.

were gradually quenched upon increasing the addition of  $\text{CrO}_4^{2-}$  or  $\text{Cr}_2\text{O}_7^{2-}$  and differences in the sensing efficacy of  $\text{CrO}_4^{2-}$  can be resolved between the two Th-MOFs. Specifically, the addition of  $50 \mu\text{M}$   $\text{CrO}_4^{2-}$  resulted in 88.1% and 58.0% decreases in the luminescence intensities of **Th-BCTPE-1** and **Th-BCTPE-2**, respectively, suggesting that **Th-BCTPE-2** is more robust against quenching than **Th-BCTPE-1** (Fig. 4a). To gain deeper insight into the quenching behaviours, the quenching constants ( $K_{\text{SV}}$ ) in the lower concentration range (0.1–1  $\mu\text{M}$ ) were derived from the Stern–Volmer (S–V) plots,  $I_0/I = 1 + K_{\text{SV}}[Q] - 1$ , where  $[Q]$  is the concentration of the quencher, and  $I_0$  and  $I$  are emission intensities before and after adding quenchers, respectively.<sup>46</sup> As shown in Fig. 4b, the S–V plots for **Th-BCTPE-1** and **Th-BCTPE-2** are linear in the range of 0.1–1  $\mu\text{M}$  and downward deviations from the linear dependence occur at higher concentrations for both materials. The  $K_{\text{SV}}$  values were calculated to be  $2.4(1) \times 10^5$  and  $1.30(7) \times 10^5 \text{ M}^{-1}$  for **Th-BCTPE-1** and **Th-BCTPE-2**, respectively, which is indicative of the higher sensing efficacy of **Th-BCTPE-1** than that of **Th-BCTPE-2**. Notably, the  $K_{\text{SV}}$  of **Th-BCTPE-1** for  $\text{CrO}_4^{2-}$  is the second highest among all MOF-based chemosensors (Table S2†).<sup>47</sup> The limits of detection (LODs) were calculated to be 9.0 and 159 nM ( $3\sigma/k$ ) for **Th-BCTPE-1** and **Th-BCTPE-2**, respectively, which agree well with that of  $K_{\text{SV}}$  (Table S3†).<sup>48</sup>

Aligned with the  $\text{CrO}_4^{2-}$  sensing, the sensing capacities of **Th-BCTPE-1** and **Th-BCTPE-2** for  $\text{Cr}_2\text{O}_7^{2-}$  were also investigated, showing that their difference in  $\text{Cr}_2\text{O}_7^{2-}$  sensing efficacy



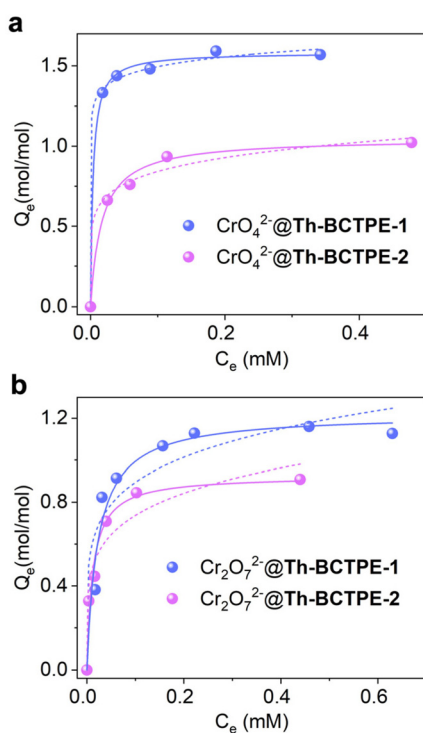
**Fig. 4** (a) The quenching rates of **Th-BCTPE-1** and **Th-BCTPE-2** as a function of  $\text{CrO}_4^{2-}$  concentration. (b) Stern–Volmer plots of luminescence quenching of **Th-BCTPE-1** and **Th-BCTPE-2** in  $\text{CrO}_4^{2-}$  solution. (c) The quenching rates of **Th-BCTPE-1** and **Th-BCTPE-2** as a function of  $\text{Cr}_2\text{O}_7^{2-}$  concentration. (d) Stern–Volmer plots of luminescence quenching of **Th-BCTPE-1** and **Th-BCTPE-2** in  $\text{Cr}_2\text{O}_7^{2-}$  solution.

is relatively trivial in the concentration range of 0–50  $\mu\text{M}$  (Fig. 4c). However, the quenching rate of **Th-BCTPE-1** is much faster than that of **Th-BCTPE-2** in a lower concentration range as shown in (Fig. S6<sup>†</sup>), suggesting that **Th-BCTPE-2** is more quenching-resistant to  $\text{Cr}_2\text{O}_7^{2-}$  than **Th-BCTPE-1**. In a similar vein, the quenching constants for  $\text{Cr}_2\text{O}_7^{2-}$  were acquired from the S–V plots, showing a higher  $K_{\text{SV}}$  ( $4.6(3) \times 10^5 \text{ M}^{-1}$ ) of **Th-BCTPE-1** than  $2.222(9) \times 10^5 \text{ M}^{-1}$  of **Th-BCTPE-2**. **Th-BCTPE-1** again exhibited a higher sensing efficacy for  $\text{Cr}_2\text{O}_7^{2-}$  than **Th-BCTPE-2** (Fig. 4d). Moreover, the LODs for  $\text{Cr}_2\text{O}_7^{2-}$  were derived from the linear regression analysis and calculated to be 4.6 and 94 nM for **Th-BCTPE-1** and **Th-BCTPE-2**, respectively, whose trend agrees well with those for  $\text{CrO}_4^{2-}$ .

Besides the quenching efficacies, the adsorption capacities of **Th-BCTPE-1** and **Th-BCTPE-2** for Cr(vi) oxyanions were also comparatively investigated. Scanning electron microscopy and energy dispersive X-ray spectroscopy (SEM-EDS) revealed that Cr was homogeneously dispersed on the surface of  $\text{CrO}_4^{2-}$ -

and  $\text{Cr}_2\text{O}_7^{2-}$ -soaked crystals (Fig. S7<sup>†</sup>). Moreover, the sorption isotherms indicated that both **Th-BCTPE-1** and **Th-BCTPE-2** can capture  $\text{CrO}_4^{2-}$  and  $\text{Cr}_2\text{O}_7^{2-}$  anions. The uptake of anionic  $\text{CrO}_4^{2-}$  and  $\text{Cr}_2\text{O}_7^{2-}$  by neutral **Th-BCTPE-1** and **Th-BCTPE-2** could be largely attributed to the electrostatic interactions between cationic SBUs and anions.<sup>38,49,50</sup> The sorption isotherms can be better fitted with the Langmuir model than the Freundlich model (Table S4<sup>†</sup>). Notably, **Th-BCTPE-1** featured higher maximum uptake capacities of  $\text{CrO}_4^{2-}$  and  $\text{Cr}_2\text{O}_7^{2-}$  (1.58 and  $1.05 \text{ mol mol}^{-1}$ ) than those of **Th-BCTPE-2** (1.22 and  $0.93 \text{ mol mol}^{-1}$ ) (Fig. 5a and b). This observation appears to be counterintuitive at first glance since **Th-BCTPE-2** has a greater void volume and a larger BET surface area than those of **Th-BCTPE-1**. However, a recent study by us revealed that the high porosity of MOFs cannot guarantee high adsorption capacity.<sup>51</sup> Since **Th-BCTPE-1** incorporates fewer solvent species, its open void is more accessible than **Th-BCTPE-2**, which renders **Th-BCTPE-1** a more efficient adsorbent for both  $\text{CrO}_4^{2-}$  and  $\text{Cr}_2\text{O}_7^{2-}$ .

Elucidating the correlation between the structure of MOF and the sensing efficacy of Cr(vi) oxyanions is essential to the rational design of new chemosensors.<sup>52–54</sup> Therefore, structural and photophysical parameters, including void volumes, BET surface areas, PLQYs,  $K_{\text{SV}}$  values, and LODs, as well as the maximum adsorption capacities, are summarized in Table 1. As seen, the porosity, surface area, and PLQY are not necessarily relevant to the sensing efficacy. In contrast, the trend of maximum adsorption capacity is in line with that of  $K_{\text{SV}}$ , implying that the sensing efficacy could be more strongly associated with Cr(vi) adsorption. Indeed, additional insights into the correlation between the sensing efficacy and sorption capacity can be derived from the S–V plots extending to higher a concentration range (Fig. S8<sup>†</sup>). For both  $\text{CrO}_4^{2-}$  and  $\text{Cr}_2\text{O}_7^{2-}$  sensing, **Th-BCTPE-1** showed saturation plateaux at higher concentrations than **Th-BCTPE-2**. These observations jointly suggest that the luminophores of **Th-BCTPE-1** are more accessible to  $\text{CrO}_4^{2-}$  and  $\text{Cr}_2\text{O}_7^{2-}$  than those of **Th-BCTPE-2**, which could be largely ascribed to the higher adsorption capacities of **Th-BCTPE-1** for Cr(vi) oxyanions. In addition, a UV-vis absorption spectroscopy study revealed that the spectra of  $\text{K}_2\text{CrO}_4$  and  $\text{K}_2\text{Cr}_2\text{O}_7$  exhibit two intense absorption peaks at 230–310 nm and 310–450 nm, both of which overlap with the excitation spectra of **Th-BCTPE-1** and **Th-BCTPE-2** (Fig. S9<sup>†</sup>). Consequently, the emission intensity reduction and eventual luminescence quenching upon titration of chromium oxyanions can be attributed to a strong inner filter effect.<sup>55–57</sup> More importantly, the overlap efficacy between the excitation



**Fig. 5** Adsorption isotherms of **Th-BCTPE-1** and **Th-BCTPE-2** for (a)  $\text{CrO}_4^{2-}$  and (b)  $\text{Cr}_2\text{O}_7^{2-}$  fitted with the Langmuir model (solid line) and the Freundlich model (dashed line).

**Table 1** The void volumes, BET surface areas, photoluminescence quantum yields,  $K_{\text{SV}}$  values, LODs, and maximum adsorption capacities of **Th-BCTPE-1** and **Th-BCTPE-2**

Products	Void (%)	BET surface area ( $\text{m}^2 \text{ g}^{-1}$ )	PLQY (%)	$\text{CrO}_4^{2-}$			$\text{Cr}_2\text{O}_7^{2-}$		
				$K_{\text{SV}}$ ( $\text{M}^{-1}$ )	LOD (nM)	$Q_{\text{max}}$ ( $\text{mol mol}^{-1}$ )	$K_{\text{SV}}$ ( $\text{M}^{-1}$ )	LOD (nM)	$Q_{\text{max}}$ ( $\text{mol mol}^{-1}$ )
<b>Th-BCTPE-1</b>	34.1	188	50.53	$2.4(1) \times 10^5$	9.0	1.58	$4.6(3) \times 10^5$	4.6	1.05
<b>Th-BCTPE-2</b>	64.7	316	53.07	$1.30(7) \times 10^5$	159	1.22	$2.222(9) \times 10^5$	94	0.93



spectrum of **Th-BCTPE-1** and the absorption spectrum of Cr(VI) is greater than that of **Th-BCTPE-2**, which makes **Th-BCTPE-1** as a more sensitive sensor for Cr(VI) detections.

The high sensing efficacy, low LOD, and excellent stability of **Th-BCTPE-1** make it well-suited for the practical application of Cr(VI) sensing (Fig. S10<sup>†</sup>). To evaluate its potential applicability to real-world conditions, the interfering effects from the environmentally relevant cations and anions were examined by dispersing finely ground **Th-BCTPE-1** in  $M(\text{NO}_3)_x$  ( $M = \text{Al}^{3+}$ ,  $\text{Cr}^{3+}$ ,  $\text{Mg}^{2+}$ ,  $\text{Ca}^{2+}$ ,  $\text{Sr}^{2+}$ ,  $\text{Cu}^{2+}$ ,  $\text{Zn}^{2+}$ ,  $\text{Mn}^{2+}$ ,  $\text{Ni}^{2+}$ , or  $\text{K}^+$ , 2 mM) and  $\text{Na}_x\text{L}$  solutions ( $\text{L} = \text{F}^-$ ,  $\text{Cl}^-$ ,  $\text{Br}^-$ ,  $\text{I}^-$ ,  $\text{IO}_3^-$ ,  $\text{NO}_3^-$ ,  $\text{HCO}_3^-$ ,  $\text{SO}_4^{2-}$ ,

$\text{BO}_3^{3-}$ ,  $\text{Cr}_2\text{O}_7^{2-}$ , and  $\text{CrO}_4^{2-}$ , 2 mM) (Fig. 6a). As shown in Fig. 6b, the luminescence of **Th-BCTPE-1** was completely quenched by  $\text{CrO}_4^{2-}$  and  $\text{Cr}_2\text{O}_7^{2-}$  with a ratio of 99.84% and 99.91%, respectively. In contrast, minor to moderate emission reductions (−3.52% to 23.62%) occurred upon immersing **Th-BCTPE-1** in interfering cation or anion solutions. These results are testaments to the excellent sensing selectivity of **Th-BCTPE-1** over the environmentally relevant cationic and anionic species.<sup>58</sup> To achieve the facile and on-site sensing of the Cr(VI) oxyanion, we further fabricated the MOF-based polyvinylidene fluoride (PVDF) membrane by dispersing finely ground **Th-BCTPE-1** in PVDF.<sup>59,60</sup> As shown in Fig. 6c, the as-prepared **Th-BCTPE-1**@PVDF membrane shows green luminescence under a 365 nm UV lamp.

Upon applying  $\text{CrO}_4^{2-}$  or  $\text{Cr}_2\text{O}_7^{2-}$  solution, a dark spot was observed and its darkness increased with the increasing Cr(VI) concentration. More strikingly, the quenched luminescence of **Th-BCTPE-1**@PVDF can be recovered by simply rinsing the membrane with DI water, indicating the excellent reusability of **Th-BCTPE-1**@PVDF.

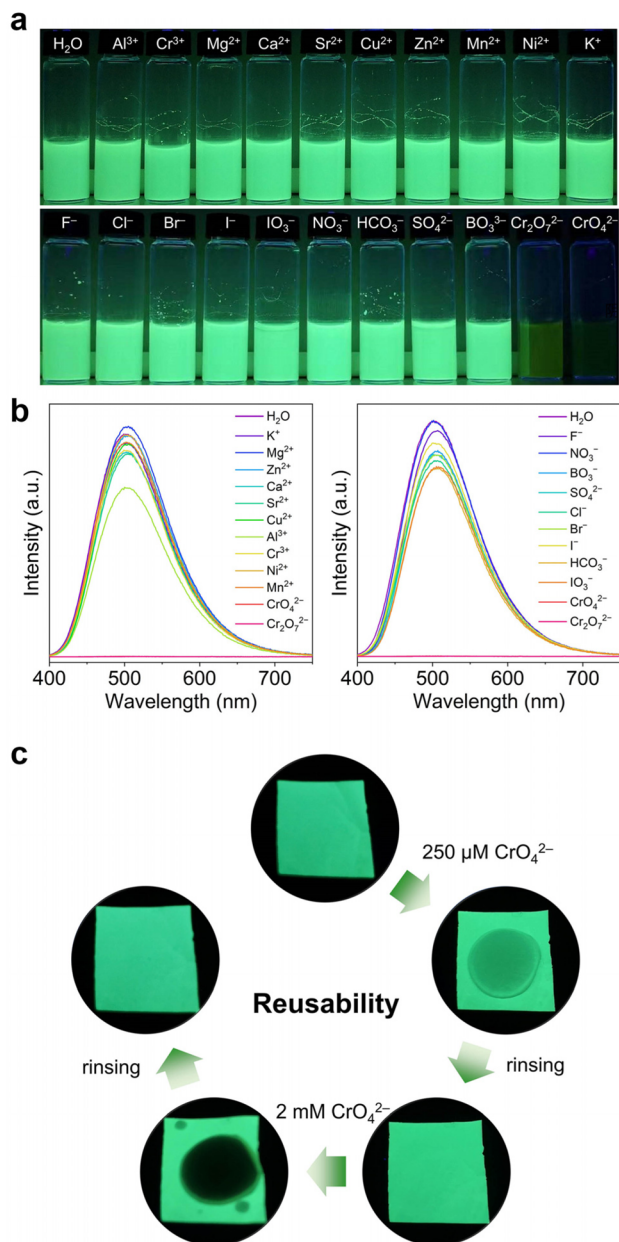
## Conclusions

In summary, LMOFs have been considered a promising material platform for sensing carcinogenic chromate and dichromate. Therefore, unravelling the correlation between the MOF structure and the sensing efficacy is essential to the rational design of high-performance MOF-based sensors. In this work, a simple synthetic modulation approach has given rise to two luminescent Th-MOFs, **Th-BCTPE-1** and **Th-BCTPE-2**, which are built from similar hexanuclear SBUs and identical BCTPE<sup>2−</sup> linkers. Such polymorphism of Th-MOFs in terms of topology allows for elucidating how the structure of MOF, in isolation, influences the sensing efficacy of Cr(VI) oxyanions. Indeed, we find that the UiO-type MOF **Th-BCTPE-1** has greater  $K_{\text{SV}}$  and lower LOD for both  $\text{CrO}_4^{2-}$  and  $\text{Cr}_2\text{O}_7^{2-}$  than the more porous **Th-BCTPE-2**. **Th-BCTPE-1** features the second highest  $K_{\text{SV}}$  of  $\text{CrO}_4^{2-}$  ( $2.84(4) \times 10^5$ ) among all MOF-based Cr(VI) sensors. Moreover, the sensing efficacies of  $\text{CrO}_4^{2-}$  and  $\text{Cr}_2\text{O}_7^{2-}$  have proved to be positively correlated with the adsorption capacity and degree of the inner filter effect; the higher ones for **Th-BCTPE-1** render its luminophores more accessible to Cr(VI) oxyanions. In contrast, other physicochemical properties, including porosity, BET surface area, and photoluminescence quantum yield, appear to play a much less significant role in determining the sensing efficacy. In addition, **Th-BCTPE-1** exhibits excellent sensing selectivity for  $\text{CrO}_4^{2-}$  and  $\text{Cr}_2\text{O}_7^{2-}$ , which is a key merit for the deployment of LMOFs in environmentally related conditions.

## Experimental section

### Synthesis

**Th-BCTPE-1.** A mixture of  $\text{Th}(\text{NO}_3)_4 \cdot 6\text{H}_2\text{O}$  (4.7 mg, 0.008 mmol), 4,4'-(1,2-diphenylethene-1,2-diyl)dibenzoic acid



**Fig. 6** (a) Photographs of **Th-BCTPE-1** immersed in different cation and anion solutions under UV irradiation. (b) Photoluminescence spectra of **Th-BCTPE-1** immersed in different cation and anion solutions. (c) Photographs of a **Th-BCTPE-1**@PVDF membrane showing its excellent reusability.

(H<sub>2</sub>BCTPE, 1.7 mg, 0.004 mmol), DMF (0.308 mL), MeOH (0.2 mL), and trifluoroacetic acid (0.04 mL) in a capped vial was heated at 80 °C for 48 h. Colourless octahedral crystals were obtained, filtered, washed with MeOH and Et<sub>2</sub>O, and dried at room temperature. The pH values before and after crystallization of **Th-BCTPE-1** were 0.91 and 5.71, respectively. Yield: 47% based on H<sub>2</sub>BCTPE. Anal. calcd for [Th<sub>6</sub>(μ<sub>3</sub>-OH)<sub>4</sub>(μ<sub>3</sub>-O)<sub>4</sub>(BCTPE)<sub>6</sub>(H<sub>2</sub>O)<sub>6</sub>](C<sub>3</sub>H<sub>7</sub>NO)<sub>18</sub>·(H<sub>2</sub>O)<sub>9</sub>, C<sub>222</sub>H<sub>268</sub>N<sub>18</sub>Th<sub>6</sub>O<sub>65</sub>, C, 47.44%; H, 4.81%; N, 4.49%. Found: C, 47.08%; H, 4.48%; N, 4.77%.

**Th-BCTPE-2.** A mixture of Th(NO<sub>3</sub>)<sub>4</sub>·6H<sub>2</sub>O (4.7 mg, 0.008 mmol), 4,4'-(1,2-diphenylethene-1,2-diyl)dibenzoic acid (H<sub>2</sub>BCTPE, 0.9 mg, 0.002 mmol), DMF (0.879 mL), and concentrated nitric acid (0.025 mL) in a capped vial was heated at 120 °C for 96 h. Colourless needle crystals were obtained, filtered, washed with MeOH and Et<sub>2</sub>O, and dried at room temperature. The pH values before and after the crystallization of **Th-BCTPE-2** were 1.12 and 5.33, respectively. Yield: 44% based on H<sub>2</sub>BCTPE. Anal. calcd for [Th<sub>6</sub>(μ<sub>3</sub>-OH)<sub>4</sub>(μ<sub>3</sub>-O)<sub>4</sub>(BCTPE)<sub>5</sub>(HCOO)<sub>2</sub>(H<sub>2</sub>O)<sub>6</sub>](C<sub>3</sub>H<sub>7</sub>NO)<sub>26</sub>·(H<sub>2</sub>O)<sub>32</sub>, C<sub>220</sub>H<sub>354</sub>N<sub>26</sub>Th<sub>6</sub>O<sub>96</sub>, C, 42.00%; H, 5.67%; N, 5.79%. Found: C, 41.51%; H, 5.33%; N, 5.99%.

### Characterization

Single-crystal X-ray diffraction data were collected on a Bruker D8-Venture single-crystal X-ray diffractometer equipped with an IμS 3.0 microfocus X-ray source (Mo-Kα radiation, λ = 0.71073 Å) and a CMOS detector. The data frames were collected using the APEX3 program and processed using the SAINT routine. The empirical absorption correction was applied using the SADABS program.<sup>61</sup> The structure was solved by intrinsic phasing with ShelXT and refined with ShelXL using OLEX2.<sup>62–64</sup> All the non-H atoms were subjected to anisotropic refinement by a full-matrix program. Contributions to scattering due to these highly disordered solvent molecules were removed using the SQUEEZE routine of PLATON.<sup>40</sup> Structures were then refined again using the data generated. Crystal data and details of the data collection are given in Table S1.† Powder X-ray diffraction (PXRD) data were collected from 2 to 40° with a step of 0.02° on a Bruker D8 Advance diffractometer with Cu Kα radiation (λ = 1.54178 Å). The calculated PXRD pattern was produced from the CIFs using the Mercury 1.4.2 program. The N<sub>2</sub> adsorption isotherms were obtained at 77 K by using a Micromeritics ASAP 2020. The freshly prepared Th-MOFs were soaked in 20 mL of DCM three times over 1 h (20 min each) and subsequently immersed in 20 mL *n*-hexane three times over 1 h (20 min each). The excitation and luminescence spectra were collected on an Edinburgh Instruments FS5 fluorescence spectrometer. The photoluminescence lifetimes and quantum yields were obtained on an Edinburgh Instruments FLS 980 spectrometer. Scanning electron microscopy (SEM) images and energy-dispersive spectroscopy (EDS) data were collected on a Zeiss Merlin Compact LEO 1530 VP scanning electron microscope with the energy of the electron beam being 15 kV. Thermalgravimetric analysis (TGA) was carried out on a NETZSCH STA 449 F3 Jupiter instrument in

the range of 30–800 °C under a nitrogen flow at a heating rate of 10 °C min<sup>-1</sup>.

## Author contributions

J. L. conceived and designed the research. Z. J. L., Y. J., and X. L. synthesized the materials and performed the luminescence study. Z. J. L. and Y. L. solved the crystal structures. X. L. W. performed the BET study. J. L., Z. J. L., J. Q., J. Q. W., Z. H. Z., M. Y. H., and L. Z. analysed the data and wrote the manuscript. All authors have given approval to the manuscript.

## Conflicts of interest

The authors declare no competing financial interests.

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