Check for updates

Strategies for industrial-grade seawater electrolysis: from electrocatalysts and device design to techno-economic analysis

Yuqing Wang, [ID] *[a,b] Feng Li,[a] Luoyin Zhao,[c] Yuchen Wang,[a] Guang Yang,[a] Jinyan Tian,[d] Shuaibing Heng,[a] Xuewen Sun,[a] Jianxu Zhao,[a] Minghua Chen [ID] *[a] and Qingguo Chen*[a]

Seawater electrolysis offers a promising route for sustainable hydrogen production, utilizing abundant seawater to meet global energy demands while addressing environmental concerns. However, challenges such as inefficiencies, high costs, and reliance on noble metal catalysts hinder its practical implementation. This review examines the fundamental mechanisms of seawater electrolysis, focusing on the hydrogen and oxygen evolution reactions (HER and OER) at industrial-scale current densities. Key strategies for catalyst design, including interfacial engineering, structural optimization, and improved mass and electron transport, to enhance efficiency and stability are discussed. Additionally, device architecture and techno-economic considerations are explored to facilitate scalable, cost-effective deployment. By providing insights into advanced materials and system innovations, this review outlines pathways for integrating seawater electrolysis into large-scale sustainable energy solutions.

[a]Key Laboratory of Engineering Dielectric and Applications (Ministry of Education), School of Electrical and Electronic Engineering, Harbin University of Science and Technology, Harbin 150080, P. R. China. E-mail: wangyq@hrbust.edu.cn, mhchen@hrbust.edu.cn, qgchen@263.net

[b]Department of Chemical and Biomolecular Engineering, College of Design and Engineering, National University of Singapore, 117576, Singapore

[c]College of Intelligent Systems Science and Engineering, Harbin Engineering University, Harbin, 150001, P. R. China

[d]State Grid Jilin Electric Power Co., Ltd Baicheng Power Supply Company, Baicheng City, Jilin Province, 137000, P. R. China

1 Introduction

Pursuing sustainable energy solutions is increasingly critical in the context of rising global energy demands and environmental concerns. Hydrogen, recognized for its exceptional potential as a renewable energy source, stands out due to its plentiful availability and high energy density. Furthermore, it offers substantial environmental benefits, producing water vapor as its only byproduct during combustion, thus eliminat-

Yuqing Wang is an associate professor at the School of Electrical and Electronic Engineering, Harbin University of Science and Technology. She obtained her Ph.D. (2020) from the Harbin Institute of Technology. From 2023, she worked as a postdoctoral researcher at the National University of Singapore. Her main research interests are focused on the development of new catalytic materials for electrolytic water splitting and plastic upcycling and recycling, and the in-depth understanding of catalytic mechanisms.

Yuqing Wang

Feng Li is a master's student at the School of Electrical and Electronic Engineering, Harbin University of Science and Technology. He graduated from East China Jiaotong University in 2019 and has been pursuing his master's degree at the Harbin University of Science and Technology since 2022. His main research interests focus on the development of new catalytic materials for the chlorine evolution reaction in water electrolysis and the in-depth understanding of catalytic mechanisms.

Feng Li

ing carbon emissions at the point of use. However, the production of hydrogen *via* conventional methods such as steam methane reforming is currently marred by high costs and significant CO_2 emissions, exacerbating the greenhouse effect. This paradox highlights the need for a transition to cleaner production techniques. Electrolysis offers a highly viable solution, especially when driven by renewable energy sources like wind, solar, or tidal power.[1] This method involves splitting water into hydrogen and oxygen using electricity and, if this electricity is derived from renewables, the entire process can be virtually emission-free. The scalability of electrolysis powered by renewable energy could radically transform hydrogen production into a truly sustainable practice. Looking ahead, integrating hydrogen production with these renewable technologies furthers the potential of hydrogen as a key component of a clean energy landscape and aligns with broader environmental goals. This strategic integration could play a pivotal role in building resilient, sustainable energy systems that cater to our growing energy needs while minimizing environmental impacts.

Hydrogen production *via* seawater electrolysis poses notable challenges alongside considerable potential.[2] Although this approach is currently less efficient and incurs higher costs compared with using purified water,[3] its appeal lies in the vast availability of seawater, which constitutes approximately 97% of the Earth's water resources. Overcoming technical barriers could unlock this abundant resource for hydrogen generation.[4] Traditional catalysts like RuO_2 and IrO_2, while effective, are both expensive and limited in supply, constraining the scalability of the process.[5] Alternatively, transition metal-based electrocatalysts offer a promising solution due to their diverse structural and compositional properties, which may improve efficiency and reduce costs.[6] These catalysts represent a promising shift from traditional noble metal catalysts, potentially catalyzing a more sustainable and economically feasible approach for hydrogen production from the vast reservoirs of seawater.

Developing noble metal-free catalysts for seawater electrolysis has become imperative to meet practical and industrial demands. Although existing catalysts perform adequately at low current densities, they fall short of the industrial benchmark, which requires exceeding 500 mA cm^{-2}, revealing a considerable performance gap.[7] Moreover, many of the reported electrocatalysts were tested at industrial-grade current densities with buffered electrolytes and alkaline environments, which only lasted a few tens of hours, while practical applications require direct electrolysis in seawater and stability for thousands of hours. While previous reviews have extensively discussed the fundamental electrochemical challenges of seawater electrolysis, including catalyst degradation, competition reactions, and scaling effects from dissolved cations such as Ca^{2+}, Na^+, and Mg^{2+}, they have largely overlooked the critical aspect of techno-economic analysis, which ultimately determines the scalability and industrial adoption of this technology. Consequently, the review not only delves into the reaction mechanisms and key challenges of hydrogen generation from seawater electrolysis at industrial-grade current densities but also uniquely integrates a techno-economic perspective, evaluating cost factors, energy efficiency, and system feasibility for large-scale implementation. By providing a comprehensive analysis of catalyst design strategies alongside seawater electrolysis device designs, it offers insights into both scientific advancements and the economic realities necessary for the widespread adoption of seawater electrolysis in hydrogen production. Finally, it discusses the future development of catalysts optimized for large-scale seawater electrolysis. The review aims to provide valuable insights for researchers, advancing the design of high-performance catalyst materials and accelerating the transition of seawater electrolysis from research to practical application.

2 Mechanisms and evaluation

2.1 Mechanisms of seawater electrolysis

The seawater electrolysis reaction generally consists of the following reactions: the OER, the HER and the chlorine oxidation

Luoyin Zhao is a Ph.D. candidate at the College of Intelligent Systems Science and Engineering, Harbin Engineering University. His main research interests are focused on the development of new energy materials related to the ocean, and research on autonomous underwater vehicles.

Luoyin Zhao

Minghua Chen has been a professor at Harbin University of Science and Technology since 2017. He received his Ph.D. from the Harbin University of Science and Technology, and was subsequently an exchange student and visiting scientist at Nanyang Technological University (NTU) from 2013 to 2016. His research interests include synthesis of 2D materials, carbon composites and heterostructure materials, and their applications for energy storage and conversion devices.

Minghua Chen

reactions (ClOR). The process of hydrogen generation through water electrolysis proceeds as follows:

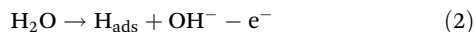$$2H_2O \rightarrow 2H_2 + O_2 \quad (1)$$

2.1.1 Mechanisms of the cathodic reaction in seawater. Under alkaline seawater conditions, the HER is relatively straightforward due to the significantly reduced concentrations of Mg^{2+} and Ca^{2+} in the electrolyte. As a result, the minimal residual Mg^{2+} and Ca^{2+} do not substantially deposit on the cathode surface, thereby preserving the accessibility of catalytic sites. Furthermore, the electric field of the cathode generates a repulsive force that minimizes the likelihood of chloride ions approaching or causing corrosion.

The occurrence of the HER at the cathode involves a two-electron reaction, comprising two steps: the Volmer, and Heyrovsky or Tafel steps, as shown in Fig. 1. Volmer step: adsorption of H_2O molecules on the active site and hydration of protons (H_3O^+); Heyrovsky step: synthesis of hydrogen atoms on the electrocatalyst surface through the electrochemical pathway of hydrogen molecules; Tafel step: hydrogen atoms synthesize hydrogen molecules directly on the electrocatalyst surface *via* the chemical pathway. The process of hydrolysis, hydroxyl adsorption (OH_{ads}), and H_{ads} exchange in alkaline seawater electrolytes take place through either the Volmer/Heyrovsky or Volmer/Tafel steps, with the reactions outlined below:

(I) Volmer step: the absence of protons leads to a scenario where the H_2O molecule interacts with electrons rather than H^+, resulting in the formation of an adsorbed hydrogen atom (H_{ads}) species on the electrocatalyst,

$$H_2O \rightarrow H_{ads} + OH^- - e^- \quad (2)$$

Fig. 1 Schematic diagram of the HER principle of alkaline seawater electrolysis.

(II) Heyrovsky step: subsequently, H_{ads} bonds with an H_2O molecule and an electron, resulting in the formation of a hydrogen molecule,
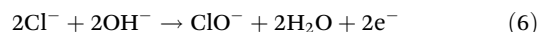
$$H_2O + H_{ads} \rightarrow H_2 + OH^- - e^- \quad (3)$$

(III) Tafel step: alternatively, a pair of H_{ads} atoms merge to create a hydrogen molecule,

$$H_{ads} + H_{ads} \rightarrow H_2 \quad (4)$$

2.1.2 Mechanisms of the anodic reaction in seawater. The competitive interaction between the OER and ClOR at the anode introduces significant challenges for sustainable hydrogen production *via* seawater splitting. Chloride oxidation is a complex process influenced by factors such as pH, temperature, and applied potential. In acidic environments, chloride ions oxidize to chlorine gas:

$$2Cl^- \rightarrow Cl_2 + 2e^- \quad (5)$$

While in alkaline conditions, they form hypochlorite:

$$2Cl^- + 2OH^- \rightarrow ClO^- + 2H_2O + 2e^- \quad (6)$$

Although the OER is thermodynamically more favorable than the ClOR, the latter, as a two-electron reaction, benefits from faster kinetics compared with the four-electron OER pathway. Under alkaline conditions, a stable potential gap of 480 mV between the OER and ClOR allows for selective seawater electrolysis. To avoid triggering the ClOR, it is crucial to maintain the overpotential of OER electrocatalysts below this 480 mV threshold at high current densities. Therefore, developing OER electrocatalysts with high selectivity, corrosion resistance, and long-term stability is essential for achieving efficient seawater splitting under industrial conditions. Metal-based electrocatalysts in seawater electrolysis are particularly vulnerable to corrosion processes, including polarization, dissolution, and hydrolysis. During seawater electrolysis, chloride ions adsorb onto and penetrate the electrocatalyst surface, forming complexes with metal ions that lead to the dissolution of active metal sites. Additionally, hydroxide ions (OH^-) in the electrolyte exacerbate this issue by promoting the formation of metal hydroxides from dissolved metal ions, further impairing the catalyst's performance and durability.

The OER represents an additional crucial half-reaction in the process of electrolyzing seawater. Occurring at the anode, this reaction differs from the HER's two-electron method by featuring a more intricate OER process, which includes a multi-stage four-electron transfer to generate O_2 at a slower rate. The common consensus is that the OER can advance through two pathways: the lattice oxygen-mediated mechanism (LOM) and the traditional adsorbate evolution mechanism (AEM).

Adsorbate evolution mechanism. In an AEM system, the oxygen evolution reaction generally involves four coordinated proton and electron transfers, with the metal center (M) serving as the active catalytic site. This process facilitates the generation of oxygen molecules from water under both acidic
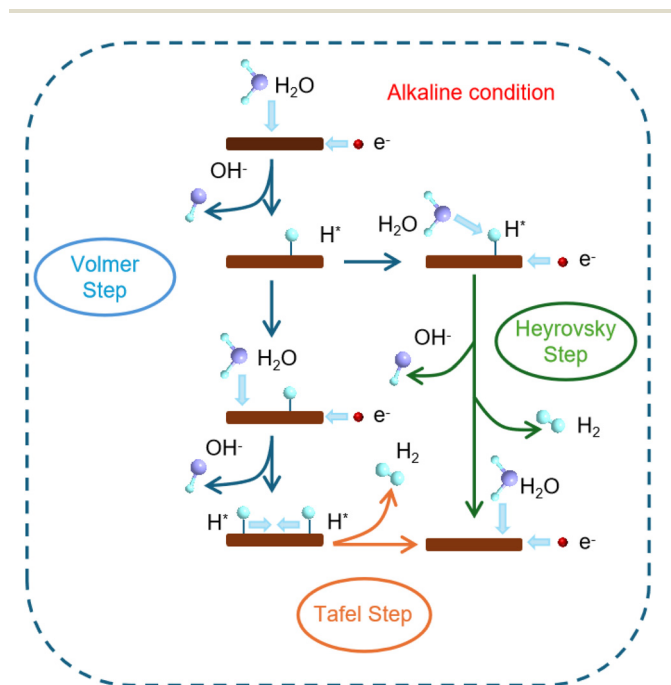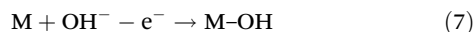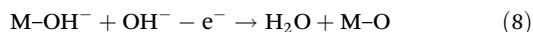
and alkaline conditions. For the OER in alkaline seawater, the reaction sequence includes the following steps:
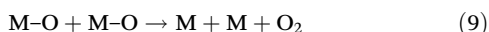
(I) The creation of an M–OH intermediate marks the initial phase in the OER procedure, focusing on OH adsorption on the electrocatalyst's surface,
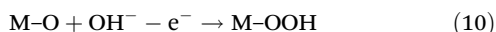
$$M + OH^- - e^- \rightarrow M\text{–}OH \tag{7}$$

(II) Continued interaction with OH^- results in the formation of M–O intermediates, which then link with H_2O and an electron.

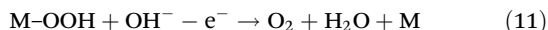$$M\text{–}OH^- + OH^- - e^- \rightarrow H_2O + M\text{–}O \tag{8}$$

(III) Subsequently, two distinct routes can lead to the creation of the O_2 molecule: the first involves the immediate attachment of two M–O species, resulting in the formation of O_2 and M active sites.

$$M\text{–}O + M\text{–}O \rightarrow M + M + O_2 \tag{9}$$

(IV) Another method involves merging M–O and OH^- to form M–OOH,

$$M\text{–}O + OH^- - e^- \rightarrow M\text{–}OOH \tag{10}$$

(V) Subsequently undergoing electron oxidation, and then utilizing a proton-coupled electron transfer technique to synthesize O_2 and active sites.

$$M\text{–}OOH + OH^- - e^- \rightarrow O_2 + H_2O + M \tag{11}$$

As per the aforementioned reaction route, comprehending the critical rate-determining stages and verifying the kinetic characterization of the OER is typically challenging, owing to the multiple intermediates referenced earlier (Fig. 2).

Lattice oxygen-mediated mechanism. Within LOM, the catalyst's lattice oxygen plays a direct role in the reaction of oxygen evolution. Lattice oxygen's role has been recently shown in
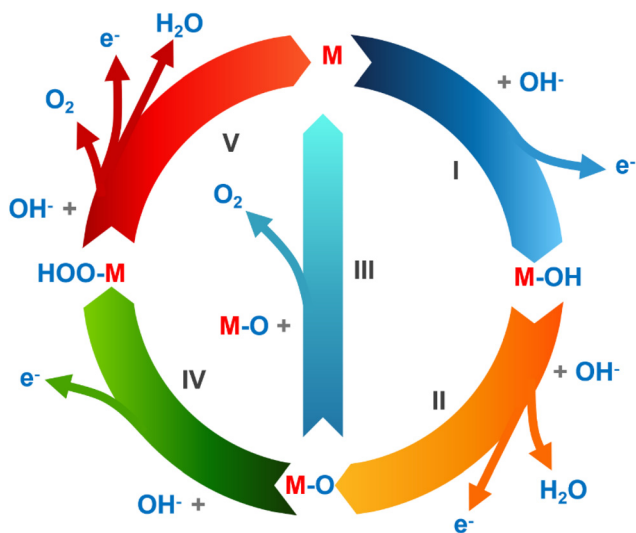

Fig. 2 Schematic diagram of the AEM process for the OER during alkaline seawater electrolysis.
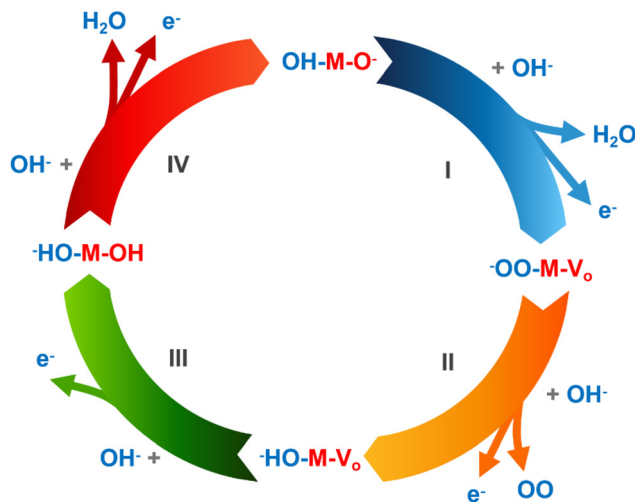

Fig. 3 Schematic diagram of the LOM process (oxygen vacancy site) for the OER during alkaline seawater electrolysis.

gas-phase catalytic oxidation processes involving alloy catalysts. Fascinatingly, this occurrence is viewed as a substitute reaction route in OER electrocatalysis, occasionally even the most advantageous (Fig. 3).

2.2 Evaluation for industrial-grade seawater electrolysis performance

The efficiency of industrial-grade seawater electrocatalysts for HER and OER is assessed through critical parameters, including overpotential (η), Tafel slope, turnover frequency (TOF), Faraday efficiency (FE), stability (both mechanical and electrochemical), and the selectivity of anodic reactions.

2.2.1 Overpotential. The initial potential is identified as the one that triggers the catalytic process, resulting in a substantial current density. Overpotential represents the variance in potential between attaining a certain current density and the thermodynamic potential of the OER/HER. An increased value of η leads to a higher consumption of electrical energy, thereby reducing the efficiency of energy conversion. An advanced electrocatalyst is capable of attaining greater current density at a reduced η value. Specifically, to align with the 12.3% photoelectrochemical water splitting efficiency, a specific current density of 10 mA cm^{-2} (η_{10}) serves as a standard for analyzing and contrasting the electrocatalysts of the HER and OER across different electrolytes. Nonetheless, η_{10} is deemed impractical for representing a catalyst's inherent activity, owing to various impacting elements like its loading mass and the extent of its active area. Additionally, for continuous hydrogen gas production in industrial electrolyzers, greater current densities (>500 mA cm^{-2}) are necessary.[7] At elevated current densities, the overpotential is determined not just by the inherent activity but also by the tested working electrode and the catalyst's surface characteristics. Calculation of the overpotential does not rely on the adjusted IR in an actual water electrolytic cell. Consequently, demonstrating IR-compensated overpotentials is crucial, particularly for self-sustain-

ing transition metal-based electrocatalysts operating at elevated current densities.

Eqn (12) outlines the operating cell voltage (V_{op}) necessary for breaking down water:

$$V_{op} = \text{IR} + V_{eq} + |\eta_c| + \eta_a \qquad (12)$$

Water electrolysis' total overpotential comprises the overpotential values of the OER (η_a) and HER (η_c), along with the voltage reduction due to system resistance, with the OER's overpotential being the primary hindrance. In this context, V_{eq} symbolizes the equilibrium potential (V_{eq} = 1.23 V), η_a and η_c denote the necessary overpotential to surmount the kinetic obstacles of the OER on the anode and HER on the cathode, respectively, while IR indicates the voltage decrease to offset the system's inherent resistance.

2.2.2 Tafel slope. The Tafel graph illustrates how the applied η varies with the current density, as inferred from the observed polarization curve. The Tafel slope's value can be ascertained by aligning the linear segment of the Tafel plot, a prevalent method for deriving the Tafel slope's value. The suggestion is to position the linear area in the low η zone, as at higher η values, the current density often strays from this linear pattern owing to the extensive bubble production.

The Tafel slope logarithmically connects overpotential with current density, expressible through eqn (13):

$$\eta = m \times \log|j| + k \left(m = \frac{2.303RT}{\alpha nF}, \ k = \frac{2.303RT \ \log|j_0|}{\alpha nF} \right) \quad (13)$$

In this context, m denotes the Tafel slope, j represents the current density, j_0 is the exchange current density, R stands for the ideal gas constant, T is the absolute temperature, n indicates the number of electrons transferred during the redox reaction, F is the Faraday constant, and α refers to the charge transfer coefficient. Within the eqn (13), a minor Tafel gradient (m) suggests the necessity of a slight overpotential shift to accommodate the swift rise in current density, thereby demonstrating superior electrocatalytic kinetics. An optimal electrocatalyst ought to possess an industrial-grade current density and a minimal Tafel slope. Furthermore, utilizing the Tafel slope aids in pinpointing potential reaction processes.

2.2.3 Turnover frequency. This refers to the quantity of reactants transformable into hydrogen/oxygen molecules every second at each catalyst's catalytic site under a specific operating potential. Additionally, TOF serves as a crucial metric for assessing an electrocatalyst's inherent activity, characterized by the speed at which the product quantity correlates with the active site count. In the context of gas evolution in electrocatalytic reactions, eqn (14) encapsulates the TOF:

$$\text{TOF} = jN_A / (nF\Gamma) \qquad (14)$$

In this context, N_A represents Avogadro's constant, j denotes the current density, n signifies the electron count transferred during the redox reaction, F is the Faraday constant, and Γ indicates the overall concentration of surface or active sites or the total atoms involved. Practically speaking,

acquiring a precise TOF value is challenging due to the unknown count of active sites. Up to this point, an appropriate technique for precisely determining the count of active sites in heterogeneous catalysis remains elusive. Certain research has straightforwardly determined TOF values using the total mass loading, presuming the electrocatalytic activity of all catalytic species. In certain instances, it's been hypothesized by scientists that solely atoms on the surface participate in the catalytic activity, with internal catalytic entities inactive, resulting in TOF measurements solely from surface catalytic locations, making precise TOF measurements a significant hurdle.

2.2.4 Faraday efficiency. This refers to how effectively the external circuit's charge is transformed into hydrogen/oxygen molecules by dividing water. Put differently, Faraday efficiency refers to the proportion between the actual and theoretical volumes of gas. Identifying the true volume of gas is achievable through water–gas displacement or gas chromatography, whereas the theoretical volume of gas is derivable from the overall charge, adhering to Faraday's electrolysis law. Typically, two techniques exist for calculating the FE of the OER. The initial approach involves estimating the theoretical oxygen production through the integration of chronopotentiometry or its analysis. An alternative technique relies on a rotating ring-disc electrode, utilizing a finely polished glassy carbon disc and a platinum ring structure. The area of the disc is aligned with the OER's identical potential window, whereas the ORR's constant potential is applied to the Pt ring. The formula for FE is represented as eqn (15):

$$\text{FE} = \frac{I_R n_D}{I_D n_R N_{CL}} \qquad (15)$$

wherein I_R and I_D represent the electrical currents emanating from the ring and the disk, in that order. In the context of the OER, 'n_R' and 'n_D' represent the electron transfer count in the ring and disk areas, correspondingly, equating to 4 by definition. N_{CL} represents the efficiency in gathering RRDE data. By-products produced in the reaction could result in Faraday losses, necessitating an ideal catalyst with a full 100% Faraday efficiency.

2.2.5 Mechanical stability. In industrial-grade seawater electrolysis, ensuring the mechanical stability of electrodes is a significant challenge that hinders the progress of hydrogen production technologies. The instability arises primarily from the interaction between electrocatalysts and their substrates, as well as with the gas bubbles formed during electrolysis. These electrodes must endure extreme conditions, including high current densities, prolonged operational cycles, elevated pressures, and high temperatures, characteristic of the harsh seawater electrolysis environment. Such demanding conditions require electrocatalysts that combine high efficiency with durability, capable of resisting mechanical degradation over time. Therefore, improving the mechanical resilience of these materials is essential for advancing reliable and efficient hydrogen production systems.[8]

In the domain of seawater electrolysis, ion polymers such as Nafion play a pivotal role at the electrode–electrolyte inter-

face, primarily serving as adhesives that anchor catalysts onto the carrier. However, their application is not without significant drawbacks. The integrity of these polymers is compromised over time due to nucleophilic attacks by water and oxygen radicals, phenomena that catalyze the degradation of the membrane electrode. This degradation is a critical concern as it leads to the weakening of the structural and functional capacity of the electrolysis system. Moreover, the issues are compounded when there is excessive polymer loading. In such scenarios, the overly thick layer of polymer can prevent effective contact between the catalyst and its carrier, eventually leading to the detachment of the catalyst. This detachment undermines the efficiency of the electrolysis process, posing a substantial barrier to the sustainable use of ion polymers in energy conversion systems. Thus, while ion polymers like Nafion are indispensable for their initial functional contributions, their long-term viability and reliability in seawater electrolysis require careful consideration and management of their application to mitigate these degradative effects.

In addition, understanding the dynamics of bubble formation at the electrocatalyst interface is crucial for optimizing the stability and efficiency of electrochemical reactions. The research findings indicate that the process of bubble formation, which encompasses nucleation, growth, and eventual separation, is a disruptive force in electrocatalytic systems. As bubbles form, they obstruct ion pathways and deactivate crucial nucleation sites, leading to a diminished active surface area and compromised electrocatalyst performance. Particularly detrimental is the peeling of the electrocatalyst from the carrier at high current densities due to strong adhesion forces exerted by the separating bubbles. Additionally, at lower current densities, the loss of active sites predominantly results from dissolution rather than mechanical forces. However, the increased flow of electrolyte, intended to enhance performance, inadvertently accelerates catalyst peeling. Therefore, enhancing the bonding strength between the electrocatalyst and its carrier is essential, as it could significantly mitigate the mechanical stresses during bubble separation and improve the overall stability and efficacy of the electrocatalytic process. This emphasizes the need for improved design and management of electrocatalyst interfaces to counteract the adverse effects of bubble dynamics on electrode stability.[9]

2.2.6 Electrochemical stability. Beyond mechanical stability, the aspect of long-term electrochemical stability holds significant value for real-world applications. Stability testing can be conducted through two techniques: accelerated cyclic voltammetry (CV) testing and either chronopotentiometry or chronoamperometric testing. During CV testing, a multitude of CV cycles, amounting to several thousand, are replicated by examining the potential of the current response. The evaluation of stability involves contrasting the polarization patterns pre and post consecutive CV cycles. A lesser rise in η at an identical current density correlates with improved stability. Additionally, the time-potential technique, also known as the time-amperometric test, is employed to verify stability at a constant potential or current density over a specified duration. The assessment of stability typically occurs at a current density of approximately 10 mA cm^{-2}. Furthermore, the industry advises using greater current densities, like 500 mA cm^{-2} or more, for conducting stability tests.[10–12]

2.2.7 Anodic reaction selectivity. The efficiency of seawater electrolysis, a pivotal process for hydrogen fuel production, is heavily impacted by the electrochemical reactions at the anode. Among these, the OER is particularly critical, as its complex four-electron transfer mechanism often makes it the rate-limiting step due to slower kinetics compared with the HER. This challenge is exacerbated by the presence of chloride ions in seawater, which promote the competing ClOR. These reactions not only reduce the selectivity of OER but also accelerate electrode corrosion, ultimately impairing the overall efficiency of the electrolysis process. Strategies to overcome these challenges include the creation of an alkaline environment and the addition of non-chlorine-containing additives such as sulfate and phosphate salts. These approaches help inhibit the ClOR and enhance the OER by reducing the local concentration of chloride ions and shifting the electrochemical potential in favor of the OER, particularly under industrial conditions where current densities are high. Moreover, developing selective anode catalysts that can alter the microenvironment near the anode presents a promising avenue to tip the balance from the ClOR towards the OER, thus optimizing the efficiency of seawater electrolysis for hydrogen production.[13]

3 Challenges faced

The majority of present-day seawater electrolysis studies take place in settings with buffered media or alkaline conditions, yet real-world use necessitates the direct electrolysis of seawater. The process of directly electrolyzing seawater encounters four additional significant obstacles.

3.1 Pollutants

Initially, a range of dissolved cations exists, including Mg^{2+}, Na^+, and Ca^{2+}, among others, as well as microbes, particulate substances, and various contaminants in natural seawater. During the electrolysis in seawater, these contaminants obstruct the electrodes, leading to poisoning and hastening the aging process of both electrodes and catalysts, thereby diminishing their longevity. For instance, commercial Pt-based electrocatalysts lose functionality in seawater within an hour due to the presence of toxic impurities in natural seawater.[14] Due to the limited buffering capacity of seawater, the pH near the anode and cathode changes significantly during seawater electrolysis. At the anode, OH$^-$ ions are consumed, leading to localized acidity, while the cathode generates OH$^-$, creating an alkaline environment. For instance, in seawater electrolysis with an initial pH ranging from 4–10, the local pH near the cathode can rise by approximately 5 pH units, while near the anode, it can drop by around 9 pH units at a current

density below 10 mA cm^{-2}.[15] The alkaline environment reduces the potential gap between the surrounding cation reduction process and HER. At the same time, in the process of cathodic precipitation, the formation of MgO, Mg(OH)$_2$, and Ca(OH)$_2$ occurs as the cathodic region's pH rises during the electrolysis phase. Over time, the build-up of these insoluble precipitates will progressively influence the reaction mechanism, hinder the diffusion of water molecules, increase the charge transfer resistance, and reduce the reaction rate of seawater electrolysis. To tackle these challenges, incorporating supporting electrolytes (such as alkali or buffers) is suggested. However, alkali additives like KOH come with a substantial cost, reaching $800 per tonne.[16] Consequently, the suggestion is for the seawater electrolysis catalyst to possess a selective permeation layer capable of eliminating contaminants from the seawater electrolysis process, selectively allowing seawater to pass through, and introducing extra electrolytes to maintain the cathode's pH balance.[2]

3.2 Competition reactions

In seawater electrolysis, the anode encounters competition between the OER and ClOR.[17] Although the OER is thermodynamically more favorable, the ClOR, as a two-electron process, exhibits a kinetic advantage over the four-electron OER, making it more competitive under certain conditions.[18] Specifically, Cl$_2$ formation is favored at pH levels below 3.0, while HClO predominantly forms between pH 3.0 and 7.5, and ClO$^-$ becomes the dominant species at pH values above 7.5.[19] Given that the OER potential diminishes as pH rises and the ClOR maintains a pH-neutral potential, chloride does not undergo substantial oxidation to ClO$^-$, with only the OER taking place in alkaline environments when η is less than 480 mV, facilitating high OER selectivity.[20] As previously stated, optimizing the thermodynamic potential difference between these reactions is essential for achieving selective OER, as it promotes OER current while preventing hypochlorite formation. Consequently, for the successful reaction of the OER in alkaline seawater electrolysis, creating an OER catalyst with high activity, capable of generating substantial current density even with minor overpotentials, is essential to circumvent the rival reaction of Cl$^-$.[2]

3.3 Peel-off of electrocatalysts

Numerous advanced seawater electrolysis catalysts have been created so far, yet the majority function at low current densities (<100 mA cm^{-2}), significantly lower than industrial standards. Creating catalysts for seawater electrolysis that possess industrial-grade current densities holds significant economic value and practical applications. Furthermore, the majority of these catalysts exist in a powdered state, and the active core might detach from the electrode upon the release of aggressive bubbles, necessitating regular substitution of catalytic substances during real-world industrial-grade current density industrial manufacturing. Consequently, employing a catalyst substrate that securely secures the catalyst and choosing electrode materials that can be easily substituted is an effective

approach. These approaches focus on securely anchoring the catalyst to prevent detachment during vigorous bubble generation and enhancing durability under high-current-density operations. For example, Luo et al.[21] reported that a hydroxide-mediated nickel-based catalyst loaded on nickel foam (h-NiMoFe) achieved high current density (1000 mA cm^{-2}) at low overpotential and could be scaled up for industrial use. Similarly, Zhang et al.[22] reported that a fluorine-doped cobalt–iron phosphide catalyst on iron foam (F-Co$_2$P/Fe$_2$P/IF) has demonstrated excellent performance and stability, highlighting the importance of integrating strong catalyst supports and replaceable electrode designs for economic and scalable hydrogen production.[23]

3.4 Bubbles

In the process of generating electrolytic hydrogen from seawater under industrial-grade current densities, numerous hydrogen gas bubbles quickly emerge. Accumulation of bubbles on the catalyst and electrolyte's contact areas significantly hinders the liquid's mass transfer, decelerates electron movement, and diminishes the count of active sites, leading to lower electrocatalytic efficiency and stability. Furthermore, the accumulation of H$_2$ and O$_2$ bubbles on the electrode creates a "bubble curtain", temporarily obstructing electrolyte access to the catalyst surface and reducing energy efficiency by about 5–10%.[24] Consequently, the primary difficulty lies in isolating the created hydrogen bubbles to preserve the electrode's catalytic efficiency during industrial-grade current density hydrogen generation in industry. It has been observed that surfaces with highly hydrophobic properties gradually emit bubbles, while electrodes with high hydrophilicity enhance the interaction between the catalyst and the liquid electrolyte. Consequently, a prevalent method for creating surfaces with both water-attracting and water-repellent qualities involves altering the texture of the catalyst electrode surface. This modification not only aids in releasing hydrogen but also expands the area exposed to the active site, thereby speeding up the electrolyte transfer mechanism.[23]

4 Design strategies for industrial–grade seawater electrolysis catalysts

In the development of industrial-grade seawater electrolysis catalysts, achieving high current densities and long-term stability is crucial. Membrane electrode assemblies (MEAs), commonly used in industrial applications, typically operate at current densities of several hundred mA cm^{-2} with catalyst loadings of 1–2 mg cm^{-2}. For proton exchange membrane water electrolyzers (PEMWEs), current densities often exceed 1.6 A cm^{-2}, while alkaline water electrolyzers (AWEs) generally reach around 0.5 A cm^{-2}. Additionally, PEMWEs are expected to maintain operational stability for over 100 000 hours in industrial settings, serving as key performance benchmarks for seawater electrolysis catalysts.[25] To meet these industrial requirements, catalyst design must focus on optimizing HER

and OER kinetics through electronic structure modifications, alloying, and heterostructure engineering to achieve low overpotentials and high reaction rates. To meet these industrial requirements, interfacial engineering plays a crucial role in enhancing the catalyst–support interaction and stabilizing active sites. Optimizing mechanical structure through nanostructuring, porous architectures, and hierarchical morphologies ensures structural integrity, facilitates rapid gas bubble release, and enhances active site exposure, which is particularly important at high current densities. Efficient electron transfer paths can be achieved by tuning the electronic structure, introducing conductive supports, and leveraging alloying or defect engineering to accelerate charge transport and reduce overpotential. Meanwhile, mass transfer optimization is critical for minimizing diffusion limitations and ensuring efficient reactant delivery and product removal. By aligning catalyst design strategies with industrial benchmarks, researchers can develop seawater electrolysis catalysts that exhibit high efficiency in laboratory conditions while also demonstrating practical viability for large-scale hydrogen production.

4.1   Interfacial engineering

Catalysts play a vital role in enhancing the efficiency of seawater electrolysis through interfacial engineering, as illustrated in Fig. 4a. Recent progress in electrocatalytic materials has been notably advanced by innovative energy research. Jiang et al.[26] fabricated NiFe LDH/FeOOH heterostructure nanosheets through electrodeposition on foam nickel–iron substrates; they have not only achieved a robust structure but have also significantly enhanced seawater electrolysis performance. This novel electrocatalyst exhibits remarkable activity and stability, key attributes for the development of active NiOOH materials. The strong interactions within the heterostructure facilitate improved electrochemical processes essential for seawater electrolysis. In the research conducted by Xiu

et al.,[27] a novel approach for improving seawater electrolysis performance is detailed, centering on the use of multi-level hollow MXene as a support material for platinum catalysts. This method notably enhanced platinum utilization, which is critical in applications where cost and material efficiency are paramount. By optimizing the catalyst structure at the nanoscale, the study achieved not only increased conductivity but also improved hydrophilicity and anti-aggregation characteristics, ensuring the catalyst's stability and effectiveness across varying pH conditions. The unique synergistic interface between the ultrafine platinum particles and the MXene structure significantly boosted charge transfer rates, facilitated hydrogen and water adsorption activation, and enhanced both intermediate hydrogen binding and ion/mass exchange. These improvements collectively contributed to a superior seawater electrolysis process, particularly in low-platinum catalyst systems where maximizing active site availability and reaction dynamics is crucial for seawater electrolysis performance.[28]

4.2   Mechanical structure

In the realm of industrial-grade electrolyzed seawater, the majority of catalysts are commonly fashioned into thin films, necessitating interaction with the current collector. As a result, precise assembly is crucial for each compact catalyst unit. The mechanical structure of the catalyst, spanning nano, micro, and macro-scale features, significantly influences channel dimensions and the extent of active site immersion in the electrolyte (Fig. 4b). Augmenting the mechanical configuration of the catalyst can elevate its efficacy in water electrolysis.

Enhancing the transport potential in industrial water electrolysis requires optimized channel designs that facilitate efficient electrolyte flow and bubble removal. Yu et al.[29] recently proposed a Ni$_2$P nanowire catalyst grown on a Ni foam substrate, showcasing remarkable electrocatalytic activity and stability in alkaline environments. This was attributed to efficient electron transfer and the robust integration of the
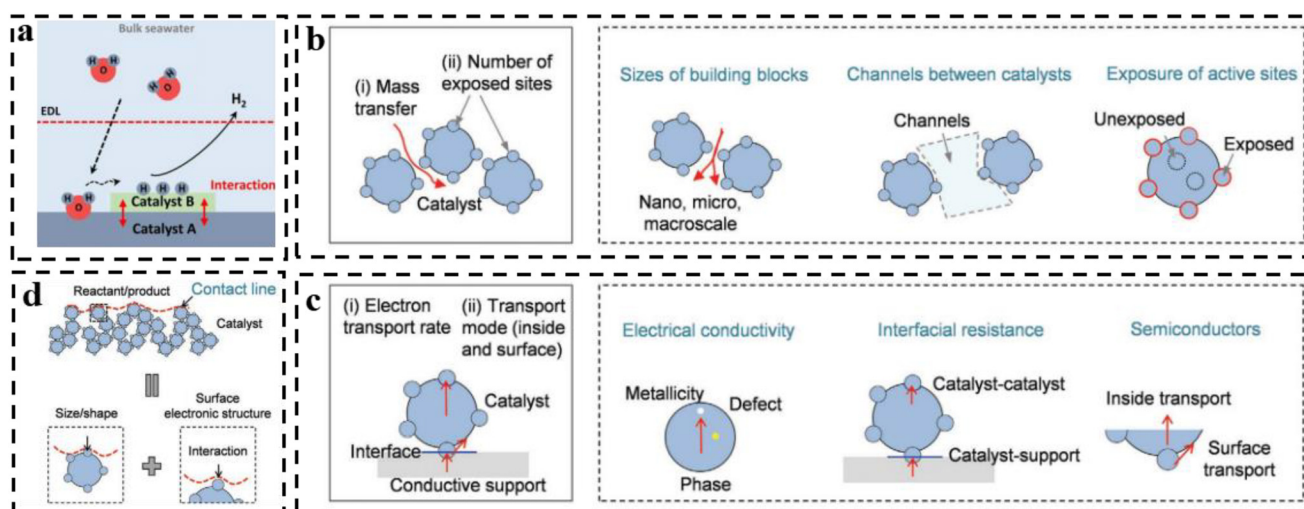

Fig. 4   (a) Interfacial engineering; (b) mechanical structure; (c) electron transfer; (d) mass transfer.[31]

nanowire array with the substrate. The Ni_2P nanowire array's "super gas-permeable" surface ensured effective hydrogen release even at high current densities exceeding 1000 mA cm^{-2}. By maximizing the exposure of catalytic sites to the electrolyte, this design enhances efficiency and provides abundant active sites for improved performance. Similarly, King et al.[30] developed an economical cobalt phosphide catalyst, scaling it from a 1 cm^2 laboratory setup to an 86 cm^2 commercial PEM electrolyzer. Using a two-step synthesis on high-surface-area carbon supports, CoP integrates seamlessly into industrial PEM electrolysis systems. Under identical conditions (400 psi, 50 °C), CoP achieved superior performance with a current density of 1.86 A cm^{-2} and sustained hydrogen production for over 1700 hours, significantly lowering material costs compared with platinum-based alternatives.[31]

4.3 Electron transfer path

Reducing resistance by optimizing electron transfer pathways is a key strategy to improve catalyst efficiency in industrial seawater electrolysis. This can be achieved by enhancing electron transport rates within the catalyst and modifying the electron transfer mechanism at the interface between the conductive carrier and the catalyst–electrolyte (Fig. 4c). Employing highly conductive materials, such as carbon, metals, or metal compounds, as catalysts or support structures can significantly boost catalytic efficiency on an industrial scale. Techniques like phase engineering and defect engineering are commonly used to adjust material conductivity. Xu et al.[32] reported a carbon-doped nanoporous cobalt phosphide ($C–Co_2P$) electrocatalyst synthesized via electrochemical treatment for the hydrogen evolution reaction. This $C–Co_2P$ catalyst achieved an overpotential of 30 mV at 10 mA cm^{-2} in 1 M KOH, exhibiting excellent stability and industrial-grade current density in artificial alkaline seawater electrolytes. Experimental results and DFT calculations revealed that carbon doping tailored the electronic structure of Co_2P, enabling the formation of $C–H_{ads}$ intermediates through the H-transfer pathway. This enhanced the HER by promoting water dissociation and hydrogen desorption. In artificial alkaline seawater containing NaCl, $MgCl_2$, and $CaCl_2$, the $C–Co_2P$ catalyst demonstrated exceptional activity and stability at industrial current densities, achieving an overpotential of 192 mV and a current density of 1000 mA cm^{-2}. These results surpass those of commercial Pt/C catalysts, highlighting the potential of $C–Co_2P$ for industrial-scale applications.

To enhance catalyst performance in industrial electrolysis, optimizing electron transfer at both catalyst–support and catalyst–catalyst interfaces is essential. The interaction between the catalyst and its support material significantly affects electron mobility, influencing interface resistance. Reducing this resistance requires careful consideration of catalyst thickness or particle size. Additionally, appropriate alignment of small structural units within the catalyst is critical to ensure efficient electron transfer over short distances. Yan et al.[33] recently developed a method for electro-synthesizing metal hydroxides/oxides using reducing acidic anions on various substrates. On

graphite, the introduction of nitrate ions into the graphene layer markedly improved the electrodeposition–support interface, resulting in enhanced loading capacity and improved hydrophilic/hydrophobic properties. In oxygen evolution electrocatalysis, co-depositing nanocrystalline cerium dioxide and amorphous nickel hydroxide on graphite achieved a low overpotential of 177 mV at 10 mA cm^{-2}. Furthermore, this composite catalyst demonstrated exceptional durability, maintaining performance at an industrial-grade current density of 1000 mA cm^{-2} for over 300 hours, underscoring its potential for industrial-scale applications.[31]

4.4 Mass transfer path

Enhancing the mass transfer process in low-dimensional catalysts is an effective strategy to improve catalytic performance in industrial seawater electrolysis, as illustrated in Fig. 4d. These catalysts influence the concentration of reactants and products at the catalytic interface, modifying the local chemical potential and boosting efficiency. Liu et al.[34] demonstrated this concept using a $NixFe_{1-x}$ alloy nanocone array with a thin NiO/$NiFe(OH)_2$ layer (~2 nm) to facilitate reactant transport. Finite element analysis revealed that the high-curvature tips of the nanocones significantly amplify the local electric field, increasing the concentration of hydroxide ions at active sites by an order of magnitude. This enhancement led to a 67% improvement in intrinsic oxygen evolution reaction activity at 1.5 V. Experimental data showed that the optimized nickel–iron alloy nanocone electrodes achieved overpotentials of 190 mV at 10 mA cm^{-2} and 255 mV at 500 mA cm^{-2}. Electrochemical surface area calibration further confirmed the superior performance of these nanocone electrodes compared with leading OER electrocatalysts. Additionally, experiments comparing samples with varying tip curvature radii reinforced the role played by tip-enhanced local electric fields in improving mass transfer, highlighting their potential for industrial applications.[31]

5 Electrocatalysts for industrial-grade seawater electrolysis

Industrial-grade seawater electrolysis catalysts are currently available for OER, HER, and bifunctional catalysts, and each of these can be categorized into noble metal catalysts, transition metal compounds, and non-metallic catalysts (Fig. 5).

5.1 OER electrocatalysts

5.1.1 Nobel metal catalysts. Nobel metals such as iridium and ruthenium are prized for their stability and performance in anode electrocatalytic water splitting, making them popular in industrial seawater electrolysis. Kang et al.[7] reported an efficient and stable seawater electrocatalyst, RuMoNi. They revealed that the catalyst consists of a row of nearly parallel nanorods uniformly coated with nanoparticles on the surfaces. After the electrochemical oxidation process, the RuMoNi electrocatalyst retains the nanorod morphology with a porous
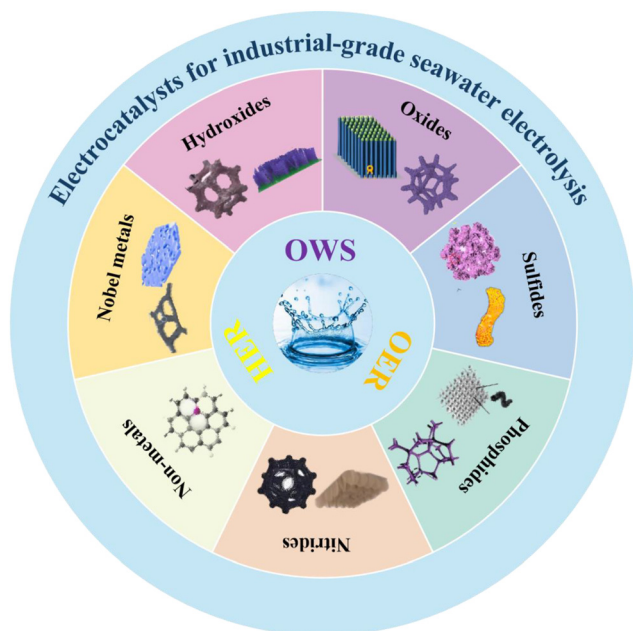
Fig. 5 Electrocatalysts for industrial-grade seawater electrolysis, including catalysts applied to the HER, OER and OWS (overall water splitting).

surface. Furthermore, high-resolution transmission electron microscopy images showed that the electrocatalyst is composed of Ni_4Mo (Fig. 6a–c). RuMoNi is emerging as a highly efficient and enduring electrocatalyst for seawater electrolysis, showcasing remarkable selectivity towards the OER alongside exceptional long-term stability. MoO_4^{2-} species formed on the

surface of RuMoNi effectively repel Cl^- ions without inducing electrode corrosion, ensuring sustained performance. The electrocatalyst demonstrates stable performance for over 3000 hours at a current density of 500 mA cm^{-2}, with minimal degradation, indicating negligible voltage increases over a projected 10-year period. Data further show that RuMoNi achieves nearly 100% OER selectivity in a 1.0 M KOH + seawater electrolyte. Additionally, an anion exchange membrane (AEM) electrolyzer using RuMoNi as the catalyst delivers seawater electrolysis at 1000 mA cm^{-2} with an impressive energy conversion efficiency of 77.9% at 1.72 V. In addition, Rossi *et al.*[35] designed a 5 cm^2 electrolyzer with iridium-coated titanium felt anodes and Pt/C-coated carbon cloth cathodes, both with a serpentine flow design. Humid air is supplied to the anode while liquid saltwater is used at the cathode, effectively minimizing chlorine buildup and enhancing overall efficiency. With a proton exchange membrane (PEM), chlorine generation is minimized, achieving oxygen evolution efficiencies of 100 ± 1% and 96 ± 2% in simulated brackish and seawater conditions. When fed with deionized water, the steam-fed anode structure matches the performance of traditional PEM electrolyzers, reaching up to 1 A cm^{-2} (Fig. 6d).

5.1.2 Transition metal compounds

Transition metal oxides. Owing to their affordability, plentiful presence, and superior resistance to corrosion, transition metal oxides are extensively utilized as OER electrocatalysts in marine settings. In the course of the electrocatalytic procedure, they typically transform into hydroxyl oxides or hydroxides, thereby demonstrating superior OER efficiency. Significantly, recent studies have demonstrated that FeNi oxides excel in oxygen precipitation reactions during seawater electrolysis at
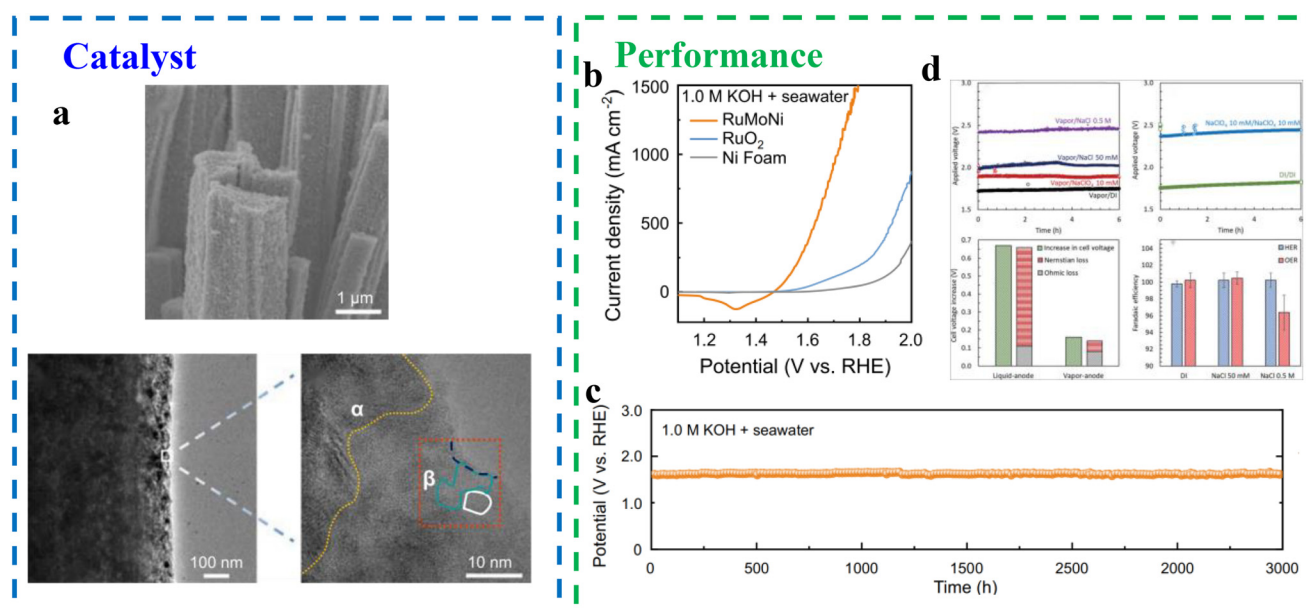

Fig. 6 (a) SEM and TEM images of the as-prepared RuMoNi electrocatalyst.[7] (b) Polarization curves of RuMoNi, RuO$_2$, and Ni Foam in a 1.0 M KOH + seawater electrolyte at a scan rate of 5 mV s^{-1} with 85% iR correction.[7] (c) Durability test for the OER for over 3000 h recorded at a current density of 500 mA cm^{-2} in a 1.0 M KOH + seawater electrolyte.[7] (d) The electrochemical performance for the steam-fed anode and liquid-fed anode electrolysis cells.[35]
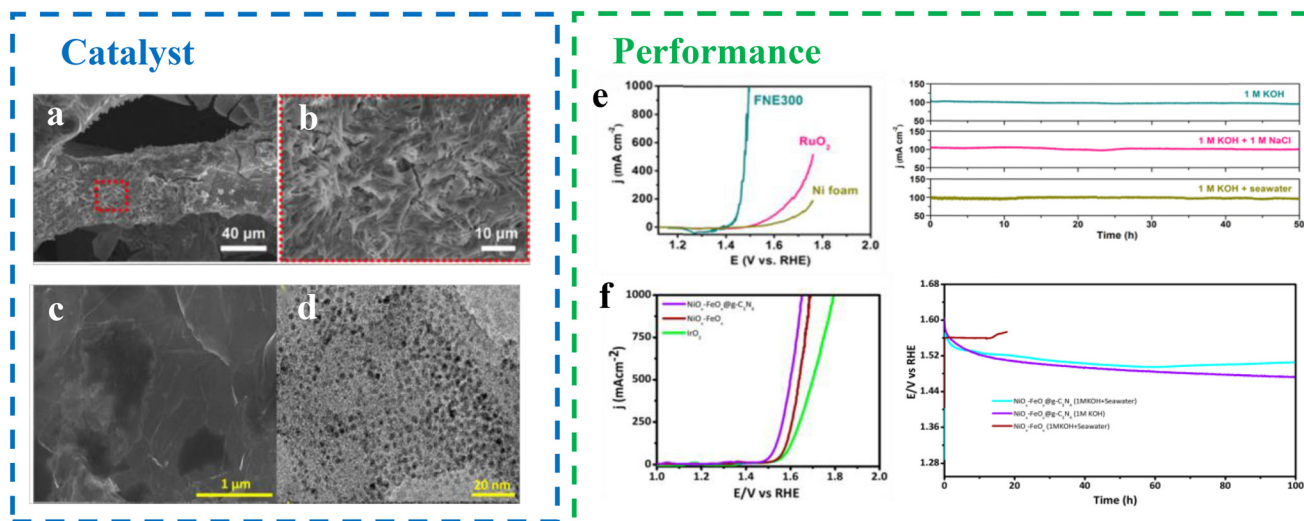
Fig. 7 (a and b) SEM and further magnified images of FNE300.[36] (c) SEM image of NiO_x–FeO_x@g-C_3N_4.[39] (d) TEM image of NiO_x–FeO_x NCs.[39] (e) Electrochemical performance of the FNE300.[36] (f) Electrochemical performance of the NiO_x–FeO_x@g-C_3N_4.[39]

elevated current densities. Li *et al.*[36] suggested an uncomplicated dipping-and-heating technique for creating Fe_2O_3/NiO heterogeneous catalysts on nickel foam (FNE300) (Fig. 7a and b). Utilizing nickel foam as the catalyst's nickel source guaranteed substantial solid interaction between the substrate and the catalyst, leading to enhanced stability and rapid electron movement. Additionally, the electrode's porous and water-attracting nature aided in electrolyte diffusion and O_2 release. The interaction between NiO and Fe_2O_3 aids in charge movement within the catalytic layer and enhances the OER kinetics at the active site.[29,37,38] To attain current densities of 10 and 1000 A cm^{-2}, the synthesized FNE300 necessitates overpotential of merely 182 and 267 mV in 1 M KOH, whereas 291 and 329 mV are needed in alkaline natural seawater and household wastewater, respectively (Fig. 7e).

Due to their unique orbital confinement and compatible electronic configuration, nickel–iron oxides are considered cost-effective and selective anodes for seawater oxidation. However, their limited stability in corrosive seawater under high current densities remains a significant challenge, hindering efficient hydrogen production *via* seawater electrolysis. Addressing this issue, Haq *et al.*[39] introduced a Ni–Fe oxide nanocluster (NiO_x–FeO_x@g-C_3N_4), supported by g-C_3N_4, as a durable and efficient anode for alkaline seawater oxidation (Fig. 7c and d). In pure seawater, the NiO_x–FeO_x@g-C_3N_4 anode achieved an industrial-grade current density of 1000 mA cm^{-2} at a low overpotential of 380 mV, maintaining this performance for over 100 hours without generating hypochlorite. The incorporation of nitrogen modifies the inherent properties of carbon, strengthening the catalyst–support interaction and enhancing electron transfer at the metal–carbon interface. This facilitates redox reactions at the electrode–electrolyte boundary. Additionally, g-C_3N_4 acts as a stabilizing agent by forming π and δ bonds, reducing interfacial resistance between the active sites and OER intermediates.[40,41] It also

protects the NiO_x–FeO_x anode from pitting and stress corrosion, significantly improving resistance to Cl^- corrosion. These features make the NiO_x–FeO_x@g-C_3N_4 system a promising solution for stable and efficient seawater electrolysis under industrial conditions. NiO_x–FeO_x NCs, backed by g-C_3N_4, exhibit greater resistance to Cl^- corrosion compared with unaltered NiO_x–FeO_x NCs. Their enhanced intrinsic activity and resistance to corrosion stem from the heterostructure's synergistic role in bolstering corrosion resistance, structural soundness, and rapid gas emission capabilities. This results in NiO_x–FeO_x@g-C_3N_4 having a larger surface area, reducing electrolyte infiltration, and controlling the chemisorption energy at the active site, thereby significantly boosting electrochemical efficiency.[42] Additionally, its industrial-grade current density maintains robust stability without noticeable degradation or hypochlorite generation during continuous electrolysis over a period of 100 hours (Fig. 7f).

Transition metal (oxygen) hydroxide. Transition metal hydroxides and hydroxide oxides are highly efficient, corrosion-resistant, and cost-effective catalysts for oxygen evolution reactions in seawater electrolysis. Their catalytic performance can be optimized by tailoring crystal structure, chemical composition, and surface modifications, leading to enhanced activity and stability. Among these, Fe–Ni-based materials have garnered significant attention for their high intrinsic activity. To achieve industrial-grade current density, strategies such as nanostructure construction have been employed to accelerate bubble release and increase exposure of active sites. Zhang *et al.*[43] developed a three-dimensional core–shell dendritic catalyst by vertically growing NiFe layered double hydroxide nanosheets (NiCo@NiFeLDH) on dendritic NiCo branches (Fig. 8a). This design offers a large surface area, efficient charge and mass transfer, and excellent corrosion resistance, making it highly active and stable in alkaline seawater. The NiCo@NiFeLDH catalyst achieved an industrial-grade current
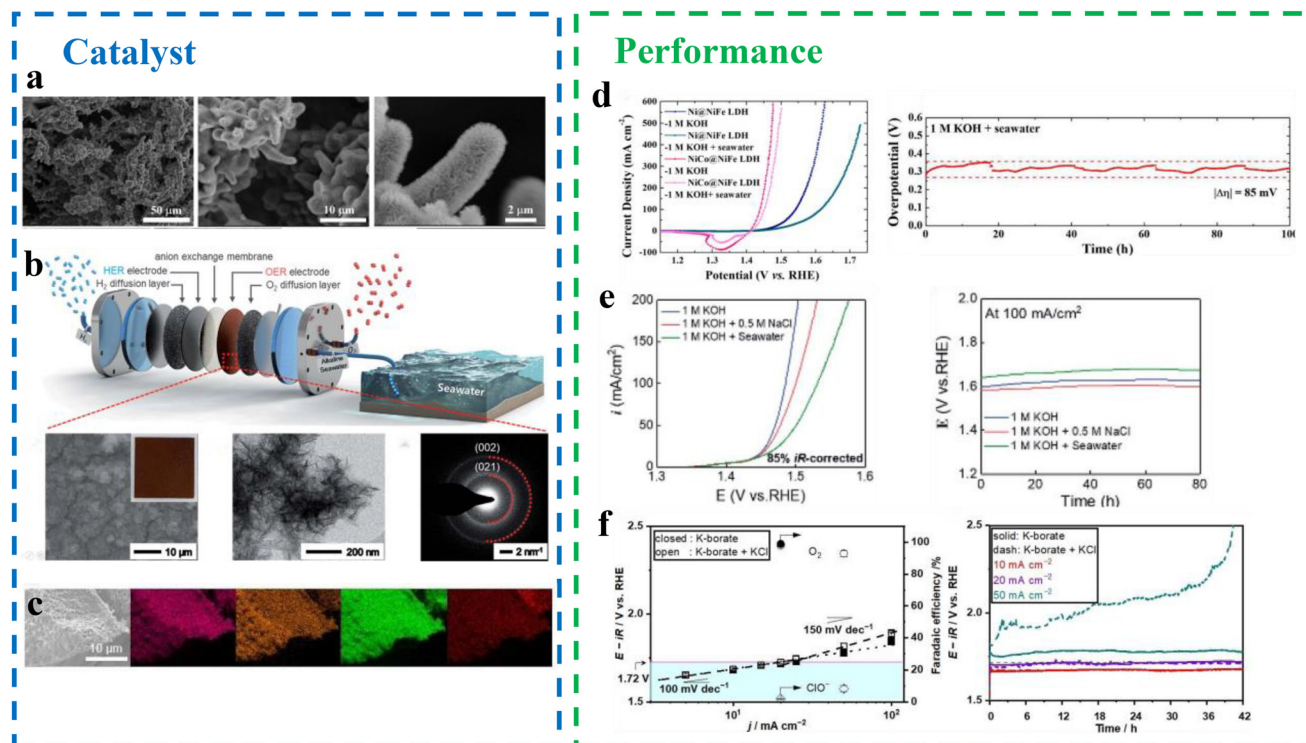
Fig. 8 (a) SEM images of NiCo@NiFe LDH.[43] (b) Schematic illustration of hydrogen production by AEM water electrolyzer in alkaline seawater.[44] SEM images with corresponding EDX mappings of (c) after OER testing of CoFeO$_x$H$_y$.[44] (d) Electrocatalytic performance of NiCo@NiFe LDH in 1 M KOH and 1 M KOH + seawater electrolytes.[43] (e) Electrocatalytic performance of the Ni-doped FeOOH in different electrolytes.[44] (f) Electrocatalytic performance of the CoFeO$_x$H$_y$/Ti.[45]

density of 500 mA cm^{-2} with a low overpotential of 266 mV. It also demonstrated nearly 100% OER faradaic efficiency, with no ClO$^-$ detected in the electrolyte after seawater electrolysis, confirming its high OER selectivity. Furthermore, the catalyst maintained stable performance with minimal overpotential fluctuations at 500 mA cm^{-2} over 100 hours of operation in seawater, highlighting its durability under industrial conditions (Fig. 8d).

Catalyst performance can be enhanced by regulating the content of doped transition metals. Park *et al.*[44] developed a nickel-doped FeOOH anode for efficient AEM alkaline seawater electrolysis (Fig. 8b and c). In half-cell tests, the Ni-doped FeOOH anode exhibited an overpotential of only 301 mV and achieved a current density of 100 mA cm^{-2} in a 1 M KOH + seawater electrolyte. The superhydrophobic nature of the Ni-doped FeOOH anode facilitates the rapid removal of generated oxygen, reducing mass transfer resistance and effectively utilizing the cell voltage. As a result, the AEM electrolyzer using this catalyst achieved a current density of 729 mA cm^{-2} with an energy conversion efficiency of 76.35% in 1 M KOH + seawater. This demonstrates the potential of Ni-doped FeOOH for high-efficiency seawater electrolysis under industrial conditions (Fig. 8e).

Iron–cobalt-based materials have recently garnered significant interest due to their high catalytic activity. Komiya *et al.*[45] developed a CoFeO$_x$H$_y$/Ti electrode with a high double-layer

capacitance, capable of stable oxygen evolution in a 0.5 mol kg^{-1} Cl$^-$ potassium borate solution at pH 9.2. This electrode features active sites composed of Co- or Fe-domains and benefits from a large reaction surface area, delivering strong OER selectivity even in Cl$^-$-rich electrolytes. The CoFeO$_x$H$_y$/Ti electrode achieved a stable current density of 500 mA cm^{-2} at a pure potential of 1.67 V *vs.* RHE at 353 K. Extensive cycling tests and long-term performance evaluations confirmed consistent OER faradaic efficiency, highlighting the electrode's durability and suitability for applications requiring high activity and stability in challenging environments (Fig. 8f).

Transition metal sulfide. Transition metal sulfides are often combined with other compounds or doped with additional metal species to enhance OER activity, enabling their application in industrial-grade seawater electrolysis. Wang *et al.*[46] developed a three-dimensional NiCoS NSAs catalyst anode for efficient and durable OER in alkaline seawater electrolysis.[47] The NiCoS NSAs catalyst employed a conductive NiCoS NSAs backbone with a large surface area, promoting the formation and activity of NiCoOOH species (Fig. 9a and b). The *in situ* generation of an amorphous S-doped NiCoOOH coating enhanced OER selectivity and provided strong corrosion resistance to chloride ions during oxygen evolution. This catalyst achieved low overpotentials of 440 and 470 mV to produce current densities of 500 and 1000 mA cm^{-2}, respectively, in alkaline natural seawater at 26 °C (Fig. 9f). Cyclic voltammetry
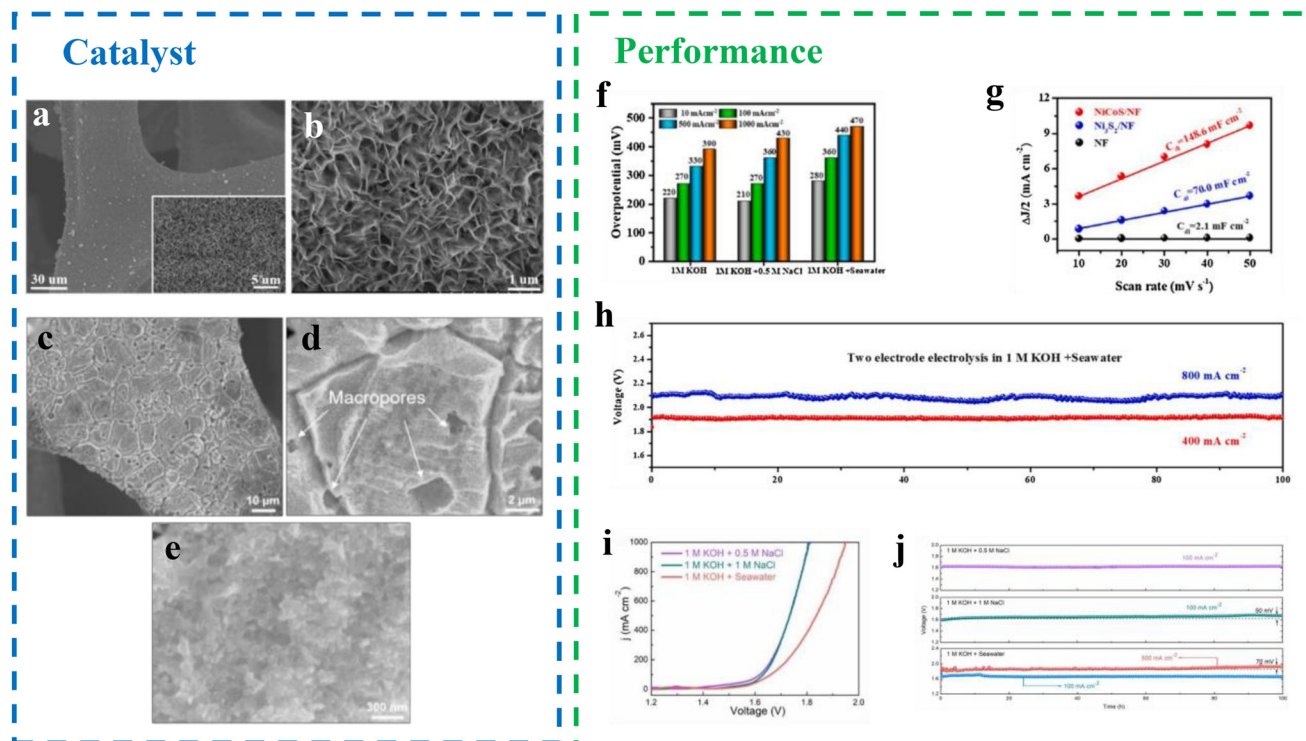
Fig. 9 SEM images of (a and b) after growth of NiCoS NSAs.[46] (c–e) S-(Ni,Fe)OOH.[48] (f–h) Electrocatalytic performance of the NiCoS NSAs catalyst.[46] (i and j) Electrocatalytic performance of the NiMoN||S-(Ni,Fe)OOH electrolyzer in different electrolytes.[48]

analysis of NiCoS catalysts with varying Co/S ratios revealed that a Co/S ratio of 3 : 10 delivered optimal OER performance (Fig. 9g). Compared with bare Ni foam and pure Ni_3S_2, the NiCoS catalysts demonstrated lower charge transfer resistance and higher electrochemically active surface area, attributed to their three-dimensional porous structure enriched with surface-active sites. The overpotential required to achieve industrial-grade current densities (up to 1000 mA cm^{-2}) remained well below the 490 mV threshold for chloride oxidation, indicating thermodynamically favorable OER activity on the NiCoS anode while suppressing unwanted chloride evolution reactions. When paired with a NiMoS catalyst for HER, this system forms a monolithic seawater electrolyzer. Operating in alkaline natural seawater, the electrolyzer achieved an industrial-grade current density of 800 mA cm^{-2} at a low voltage of 2.08 V, maintaining good durability over 100 hours of operation (Fig. 9h). This demonstrates the potential of NiCoS NSAs for industrial seawater electrolysis applications.

Sulfur doping into transition metal compounds *via* a one-step surface engineering approach significantly enhances their OER activity. Yu *et al.*[48] reported high-performance S-doped Ni/Fe(oxy)hydroxide catalysts for seawater electrolysis. This method rapidly prepares commercial nickel foams at room temperature, transforming them into catalysts with multi-layered pores and large surface areas (Fig. 9c–e). Unlike conventional electrodeposition or hydrothermal methods, this approach directly reacts with nickel foam in solution, quickly

etching a stable, hydrophilic S-doped Ni/Fe(oxy)hydroxide layer. This ensures robust contact, facilitating fast electron transfer and excellent stability. Simultaneously, sulfur is incorporated into both the surface and lattice of the Ni/Fe(oxy) hydroxide, tuning the valence states of Ni and Fe and optimizing the adsorption energy of OER intermediates, thereby boosting catalytic activity. The high density of active sites, combined with the catalyst's stability, results in outstanding OER performance in 1 M KOH + seawater electrolyte. When paired with the advanced HER catalyst NiMoN, the two-electrode electrolyzer achieves industrial-grade current densities at low voltages, demonstrating remarkable durability. Specifically, the system delivered current densities of 500 and 1000 mA cm^{-2} at low voltages of 1.837 V and 1.951 V, respectively (Fig. 9i and j). These results offer a practical and efficient solution for hydrogen production *via* seawater electrolysis under industrial conditions.

Transition metal phosphides. Transition metal phosphides, like transition metal sulfides, achieve efficient OER performance through phosphorus intercalation. Yu *et al.*[49] introduced an anion-assisted strategy to enhance electrocatalyst performance, enabling rapid screening of electrolyte additives by analyzing the correlation between chloride rejection and anionic properties (Fig. 10a–c). This approach enabled highly stable alkaline seawater electrolysis at industrial-scale current densities of 0.5 A cm^{-2} for 500 hours by incorporating phosphate ions into transition metal hydroxides as anodic electrocatalysts. Using NiFe LDH as a prototype, the study explored the
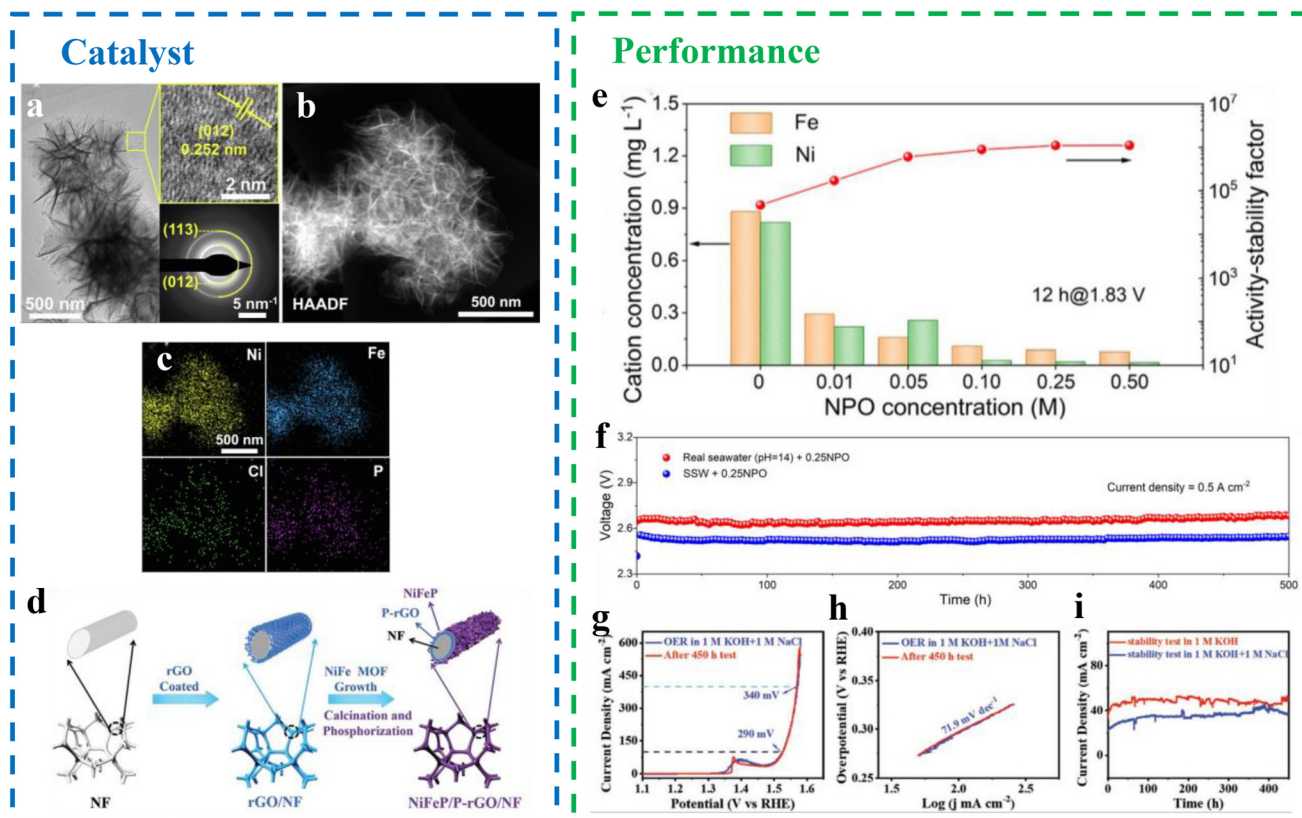
Fig. 10 (a–c) The TEM, HAADF-STEM images and the corresponding elemental mappings of NiFe LDH after stability tests in SSW + 0.5NPO.[49] (d) Schematic diagram of the fabrication (NiFeP/P-rGO/NF) process.[51] (e) The concentration of Ni/Fe ions in electrolyte and the corresponding activity–stability factor after 12 h chronoamperometry at 1.83 V with different concentrations of NPO.[49] (f) Long-term stability of NiFe LDH@Ni in SSW and real alkaline seawater with 0.25 NPO using two-electrode electrolysis cell.[49] (g–i) Electrocatalytic performance of the NiFeP/P-rGO/NF electrocatalyst.[51]

role played by anionic additives in OER catalysis within alkaline seawater electrolytes.[50] Comparative electrochemical analysis revealed that the addition of PO_4^{3-} ions significantly enhanced OER stability and faradaic efficiency compared with other anions with varying charges and radii. Increasing the concentration of PO_4^{3-} effectively suppressed metal dissolution, and the activity–stability factor improved nearly 24-fold with the inclusion of 0.25 M Na_3PO_4, highlighting the anti-corrosion properties of phosphate ions (Fig. 10e). In a three- or two-electrode system using true alkaline seawater with 0.25 M Na_3PO_4, NiFe LDH@Ni sustained operation for 500 hours at a current density of 0.5 A cm^{-2} (Fig. 10f). When paired with Pt foil, the system exhibited negligible voltage attenuation in real seawater, showcasing exceptional corrosion resistance and performance comparable to the most advanced seawater splitting systems reported in recent years. This demonstrates the potential of phosphate-assisted electrocatalysts for industrial-scale seawater electrolysis.

Transition metal phosphides can also be synthesized by combining phosphorus with transition metal compounds to achieve efficient OER performance. Qi et al.[51] fabricated nickel foam (NF)-enclosed, reduced graphene oxide (rGO)-supported subnanoscale nickel phosphide arrays (Fig. 10d). These arrays

were prepared using metal–organic framework (MOF) precursors, which provide tunable active sites and a high density of exposed catalytic sites.[52–54] The resulting NiFeP/P-doped rGO/NF (NiFeP/P-rGO/NF) catalyst demonstrated outstanding performance in seawater electrolysis. In a 1 M KOH + 1M NaCl electrolyte, the NiFeP/P-rGO/NF exhibited low overpotentials of 290 and 340 mV at current densities of 100 and 400 mA cm^{-2}, respectively (Fig. 10g), significantly below the 490 mV threshold for initiating chloride oxidation. The catalyst also showed a low Tafel slope of 71.9 mV dec^{-1}, indicating rapid reaction kinetics (Fig. 10h). Durability tests revealed no significant loss in current density after 450 hours of operation at 35 mA cm^{-2} (Fig. 10i). The excellent catalytic activity is attributed to the ultra-thin NiFeP arrays, which provide abundant active sites, and the conductive rGO network that facilitates efficient electron transfer. Additionally, the hydrophobicity of rGO ensures stability and durability in seawater, making NiFeP/P-rGO/NF a promising candidate for sustainable and efficient seawater electrolysis.

Noble metal catalysts demonstrate exceptional stability, high OER selectivity, and strong long-term performance, particularly in industrial applications. In contrast, transition metal compounds present cost-effective alternatives, offering

Table 1 Performance comparison of OER electrocatalysts

Catalyst type	Catalyst	Performance	Testing conditions	Ref.
Transition metal oxides	Fe_2O_3/NiO@NF	291 mV@1000 mA cm^{-2}/ 252 mV@1000 mA cm^{-2}	1 M KOH + seawater/1 M KOH + 1M NaCl	55
Transition metal oxides	NiO_x–FeO_x @g-C_3N_4	380 mV@1000 mA cm^{-2}	1 M KOH	56
Transition metal (oxygen) hydroxide	NiCo@NiFe LDH	222 mV@100 mA cm^{-2} 266 mV@500 mA cm^{-2}	1 M KOH + seawater	57
Transition metal (oxygen) hydroxide	Ni doped FeOOH	1.7 V$_{cell}$@729 mA cm^{-2}	AEM + 1 M KOH + seawater	58
Transition metal (oxygen) hydroxide	$CoFeO_xH_y$/Ti	1.67 V(vs. RHE)@500 mA cm^{-2}	1.0 mol kg^{-1} K-borate + 0.5 mol kg^{-1} KCl, pH 9.2 + 353 K	59
Transition metal sulfide	NiCoS NSAs @NF	360 mV@500 mA cm^{-2} 430 mV@1000 mA cm^{-2}/ 440 mV@500 mA cm^{-2}	1 M KOH + 0.5 M NaCl/1 M KOH + seawater	60
Transition metal sulfide	S doped (Ni,Fe)OOH	339 mV@500 mA cm^{-2} 378 mV@1000 mA cm^{-2}/ 398 mV@500 mA cm^{-2} 462 mV@1000 mA cm^{-2}	1 M KOH + 0.5 M NaCl/1 M KOH + seawater	48
Transition metal phosphides	PO_4^{3-} doped NiFe LDH@Ni	2.674 V(vs. RHE)@500 mA cm^{-2}	Real alkaline seawater + 0.25 M Na_3PO_4	49
Transition metal phosphides	NiFeP/P-rGO/NF	340 mV@400 mA cm^{-2}	1 m KOH + 1 m NaCl	51

enhanced corrosion resistance and significant efficiency gains through nanostructuring and doping strategies. Among them, NiFe-based materials exhibit notable intrinsic activity and stability, with modifications such as phosphate incorporation and sulfur doping effectively mitigating Cl$^-$ corrosion and optimizing performance at high current densities. The findings emphasize the critical role played by structural engineering, chemical composition tuning, and surface modifications in improving both activity and durability across all catalyst types. While transition metals provide economic advantages, noble metals ensure superior stability, making them complementary solutions adaptable to specific operational requirements. Table 1 summarizes the performance comparison of OER electrocatalysts for seawater electrolysis at industrial-grade current density.

5.2 HER electrocatalysts

5.2.1 Nobel metal catalysts. Owing to their corrosion resistance and exceptional catalytic efficiency, precious metal-based catalysts are widely utilized in the cathodic HER for industrial seawater electrolysis. Guo et al.[61] developed Ru and P co-doped NiMoO$_4$ multichannel nanorods (Ru/ P-NiMoO$_4$@NF), grown in situ on nickel foam. This catalyst enables chlorine-free hydrogen production by integrating seawater splitting with thermodynamically favorable urea oxidation (Fig. 11a). It achieved an industrial-grade current density of 3000 mA cm^{-2} at an operating voltage of 0.23 V (Fig. 11d). Similarly, Tran et al.[62] synthesized a three-dimensional nanostructure comprising single-atom rhodium and amorphous nickel hydroxide nanoparticles anchored onto copper nanowire arrays on copper foam (NiRh-CuNA/CF) (Fig. 11b). Fabricated through a straightforward electrodeposition and cation exchange process, the catalyst features a high density of active sites, a large specific surface area, and excellent electrical conductivity. The NiRh-CuNA/CF exhibited superior HER activity with overpotentials of 12 mV in freshwater and 21 mV in seawater at a current density of 10 mA cm^{-2} (Fig. 11e). It maintained excellent performance under

industrial-grade conditions, achieving 300 mA cm^{-2} at an overpotential of 155 mV and demonstrating minimal current fluctuations over 30 hours (Fig. 11f). Experimental findings revealed that the amorphous nickel hydroxide nanoparticles effectively lower water dissociation energy, accelerating alkaline seawater electrolysis. Meanwhile, single-atom rhodium serves as the active site for hydrogen adsorption and recombination, facilitating efficient hydrogen generation.

The incorporation of platinum group metals (PGMs) significantly enhances the HER activity of transition metal alloy catalysts. Yang et al.[63] synthesized a high-performance Pt/Ni–Mo electrocatalyst using a sequential reduction strategy. This catalyst consisted of PGM nanoparticles anchored on a high-surface-area, corrosion-resistant substrate designed for efficient electrolysis. The synthesis involved two key steps: first, a hydrothermal process was used to grow Pt/NiMoO$_4$ micro-pillars on nickel foam by dissolving H_2PtCl_6, $(NH_4)_2MoO_4·7H_2O$, and $Ni(NO_3)_2·6H_2O$ in an aqueous solution. Next, the Pt/NiMoO$_4$ was reduced using hydrogen gas (Fig. 11c). During this reduction, Pt nanoparticles and Ni_4Mo nanocrystals were released and immobilized on the NiMoO$_4$ substrate while maintaining the micropillar morphology. This design, with highly dispersed PGM nanoparticles and a tailored substrate, resulted in exceptional catalytic performance. The Pt/Ni–Mo catalyst achieved an overpotential of 42 mV for HER and 420 mV for OER at an industrial-grade current density of 2000 mA cm^{-2} in a 1 M KOH solution. In a brine electrolyte (1 M KOH and 0.5 M NaCl), the Pt/Ni–Mo electrocatalyst required only 113 mV to reach the same current density (Fig. 11g). The catalyst demonstrated high activity and durability across a range of challenging conditions, including strong alkaline and simulated seawater electrolytes. These characteristics highlight its potential for industrial-scale applications under harsh operating environments.

5.2.2 Transition metal compounds

Transition metal phosphides. Transition metal phosphides have garnered significant attention as efficient HER catalysts
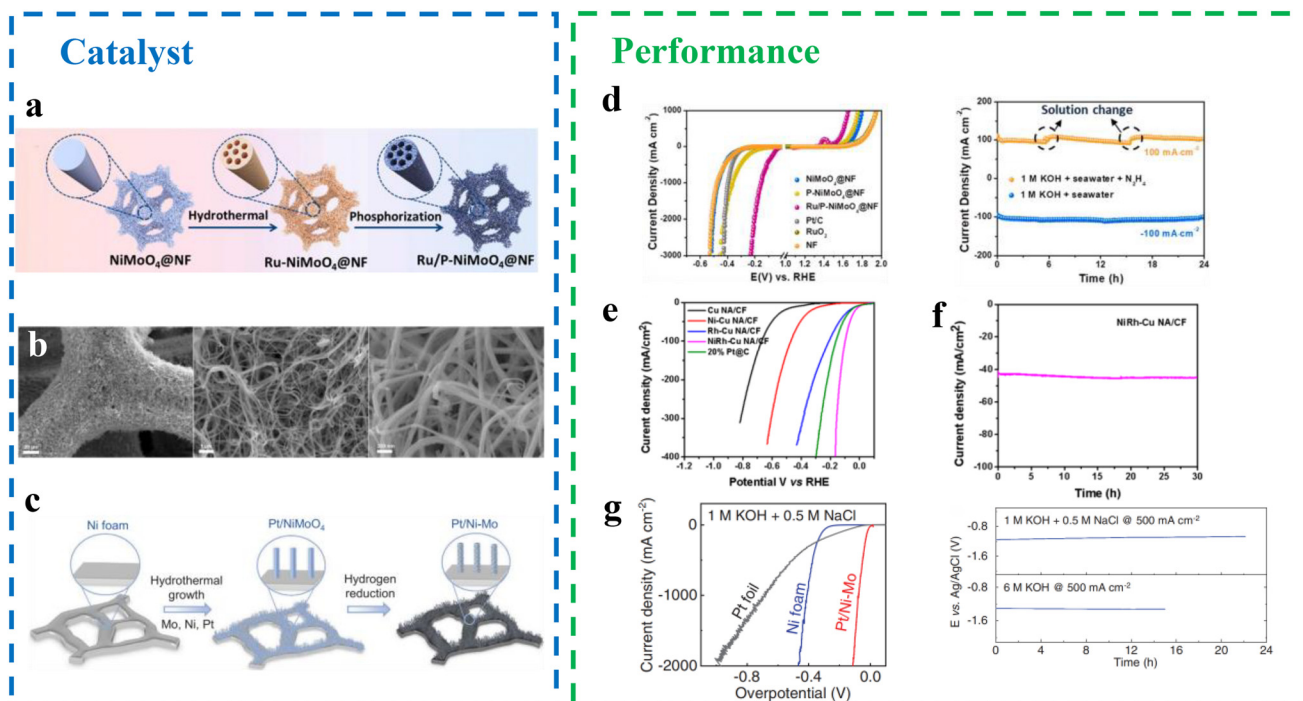
Fig. 11 (a) Schematic illustration of the formation process of Ru/P-NiMoO$_4$ @NF.[61] (b) FESEM images of NiRh-Cu NA/CF.[62] (c) Schematic of the two-step synthesis process (Pt/Ni–Mo).[63] (d) Electrochemical performance of the Ru/P–NiMoO$_4$@NF catalysts.[61] (e) LSV curves and (f) long-term stability test of NiRh-Cu NA/CF.[62] (g) Electrochemical performance of the Pt/Ni–Mo electrocatalyst.[63]

for high-current-density seawater electrolysis. Nanostructured phosphides grown *in situ* on conductive substrates such as carbon cloth (CC), carbon fiber (CF), nickel foam (NF), and carbon paper (CP) have shown remarkable potential for these applications. These substrates provide large surface areas for active site growth, enhance charge transport across electrodes, and facilitate gas discharge and reactant–product exchange due to their porous structures. Chai *et al.*[64] developed a bimetallic phosphide catalyst (Cu$_3$P-FeP@CC) grown on carbon cloth for HER at ultra-industrial-grade current densities (Fig. 12a). The catalyst demonstrated excellent HER performance, achieving low overpotentials of 338 mV and 406 mV at a current density of 1000 mA cm^{-2} in alkaline and salt-alkaline electrolytes, respectively (Fig. 12g and h), and maintaining stability for up to 400 hours. Carbon cloth, with its high electrical conductivity, elasticity, and mechanical strength, provides numerous attachment points for the catalyst due to its woven structure. When tightly bonded with the catalyst, it enhances active site exposure and electrolyte diffusion, promoting efficient HER. In this system, FeP acts as the primary hydrogen adsorption site, while Cu$_3$P adjusts the electronic structure and electron orbitals of FeP, shifting the d-band center closer to that of Pt.[65] This synergy mimics Pt-like hydrogen evolution properties, enhancing catalytic kinetics and activity. Li *et al.*[66] synthesized a P–Ni$_4$Mo catalyst grown on copper foam (P–Ni$_4$Mo/CF) (Fig. 12b–d), achieving appreciable HER performance in alkaline seawater with an overpotential of 260 mV at 100 mA cm^{-2} and 551 mV at 1 A cm^{-2}, slightly underperforming compared with the Pt/C catalyst (453 mV)

(Fig. 12i). The catalyst demonstrated strong durability, with negligible activity loss over 200 hours of testing (Fig. 12j). Yu *et al.*[67] proposed incorporating small molecules like hydrazine (N$_2$H$_4$) into alkaline-saline electrolytes to shift anodic oxidation potentials negatively. This strategy increases the potential gap between hypochlorite formation and anodic small-molecule oxidation reactions, allowing for high cathodic HER current densities without generating corrosive hypochlorite species.[68] Using bifunctional Co–Ni–P nanowires supported on carbon paper (Co–Ni–P/CP) as both cathode and anode (Fig. 12e), they achieved hydrogen production at 500 mA cm^{-2} in an alkaline brine electrolyte with a cell voltage of 0.533 V and excellent stability for 80 hours with minimal degradation (Fig. 12k). These studies underscore the potential of transition metal phosphides and advanced system designs to enhance HER performance and durability in industrial seawater electrolysis.

Recent advancements in tuning the intrinsic interfacial electronic structure of catalysts through multiphase synergistic interactions have drawn considerable attention for enhancing HER performance. Li *et al.*[69] developed a multiphase interface catalyst (MPIC), 1T$_{0.63}$-MoSe$_2$@MoP, which exhibited exceptional HER activity in both alkaline freshwater and simulated seawater electrolytes. The catalyst demonstrated current densities of 1000 mA cm^{-2} with performance comparable to Pt-based materials (Fig. 12f). This MPIC leverages multiphase synergistic intrinsic interfacial electronic structure engineering to achieve efficient HER electrocatalysis. The unique modulation of interfacial electronic structures in the 1T$_{0.63}$-
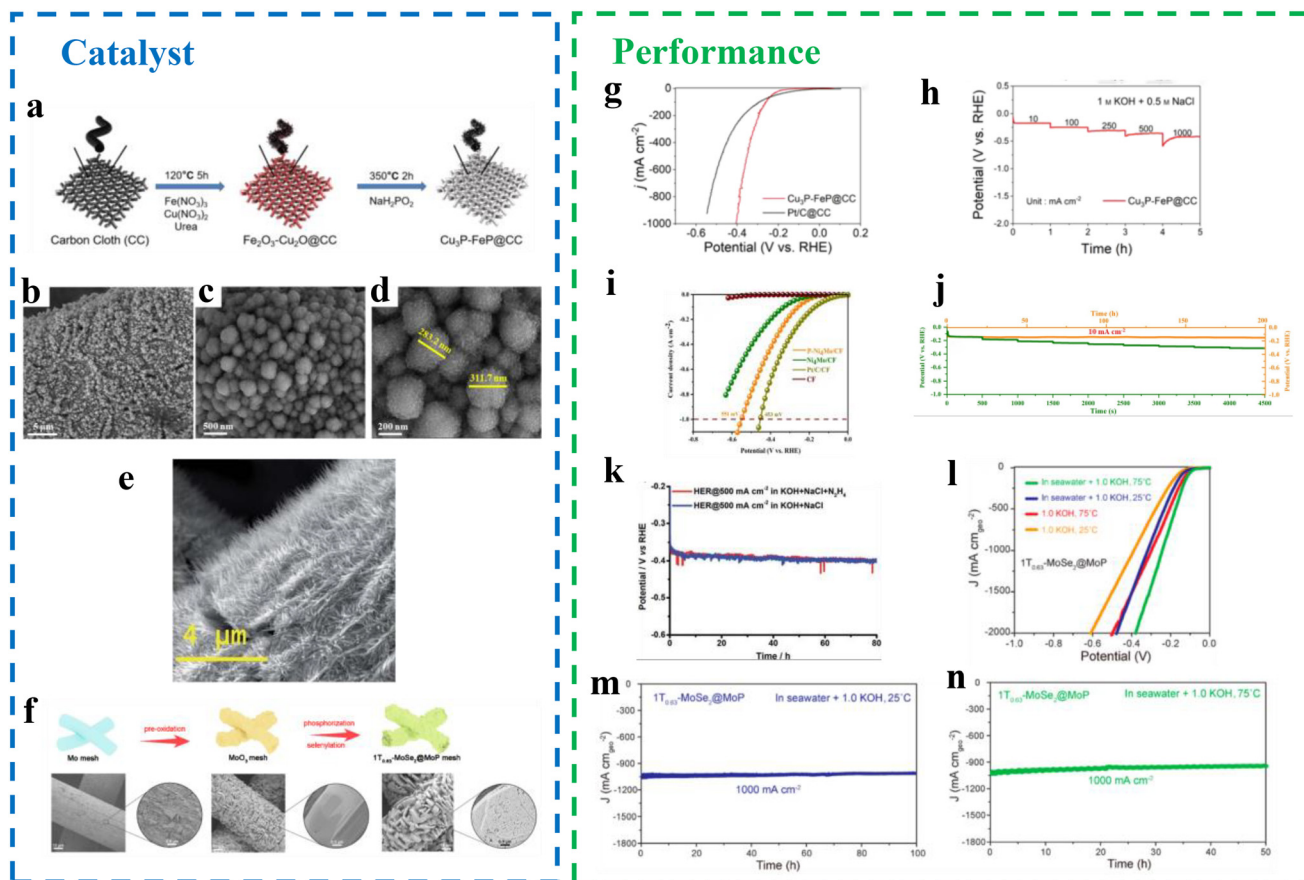
Fig. 12 (a) Illustration of preparation procedures of the Cu_3P-FeP@CC.[64] (b–d) SEM of P-Ni_4Mo/CF.[66] (e) SEM of the Co–Ni–P/CP.[67] (f) Schematic illustration of the fabrication process for self-standing $1T_{0.63}$-$MoSe_2$@MoP MPIC and corresponding SEM images.[69] (g) LSV curves, (h) Multicurrent chronopotentiometric curves of the Cu_3P-FeP@CC.[64] (i) LSV curves, (j) multipotential stability test of P–Ni_4Mo/CF.[66] (k) Chronopotentiometric curves of Co–Ni–P/CP at 500 mA cm^{-2}.[67] (l–n) Electrochemical performance of the $1T_{0.63}$-$MoSe_2$@MoP MPIC.[69]

$MoSe_2$@MoP MPIC facilitates the formation of multiple heterostructures, including $2H$-$MoSe_2$, $1T$-$MoSe_2$, and MoP phases. These heterostructures enhance charge transfer and catalysis, enabling high activity. The catalyst is self-supporting and vertically grown on industrial-grade molybdenum meshes through a straightforward solid-state synthesis process under non-sensitive conditions. The method allows for scalable preparation of large MPICs, constrained only by the dimensions of the growth chamber and the molybdenum mesh. As a result, the $1T_{0.63}$-$MoSe_2$@MoP MPIC achieved impressive current densities of 1000 mA cm^{-2} in both alkaline freshwater and artificial seawater electrolytes, requiring low overpotentials of 358 mV and 317 mV, respectively (Fig. 12l–n). This work highlights the potential of interfacial electronic structure tuning for developing high-performance, scalable HER catalysts.

Transition metal sulfide. Transition metal sulfides, known for their physical and chemical similarities to transition metal phosphides, have emerged as promising HER catalysts for high-current-density seawater electrolysis aimed at hydrogen production. Lv *et al.*[70] developed self-supported ultrathin and metastable FeS nanolayers on a three-dimensional porous iron foam (IF) substrate using a one-step mild sulfation etching

process, creating FeS@IF electrodes within just two hours (Fig. 13a). The FeS@IF electrodes demonstrated excellent and stable HER performance in various electrolytes, including alkaline simulated seawater (1 M KOH + 0.5 M NaCl), neutral electrolytes (1 M PBS), and other corrosive environments. The catalyst achieved low overpotentials of 63 mV and 78 mV to reach a current density of 10 mA cm^{-2} in 1 M KOH + 0.5 M NaCl and 1 M PBS, respectively (Fig. 13c). Furthermore, the FeS@IF electrode sustained operation at 0.2–0.4 A cm^{-2} for over 600 hours in neutral electrolyte (1 M PBS) with negligible performance degradation (Fig. 13d). The excellent durability and activity of the FeS@IF electrode can be attributed to the ultrathin, corrosion-resistant FeO_xS_y nanoscopic fan-shaped structures formed during etching. This uniform morphology enhances interfacial electron transport between active species and the substrate, exposes abundant active sites, and creates optimized channels for the rapid release of bubbles and efficient mass transfer. These properties make FeS@IF electrodes a highly effective and durable solution for HER in challenging seawater electrolysis environments.

The incorporation of heterogeneous structures between transition metal sulfides and other metallic materials can sig-
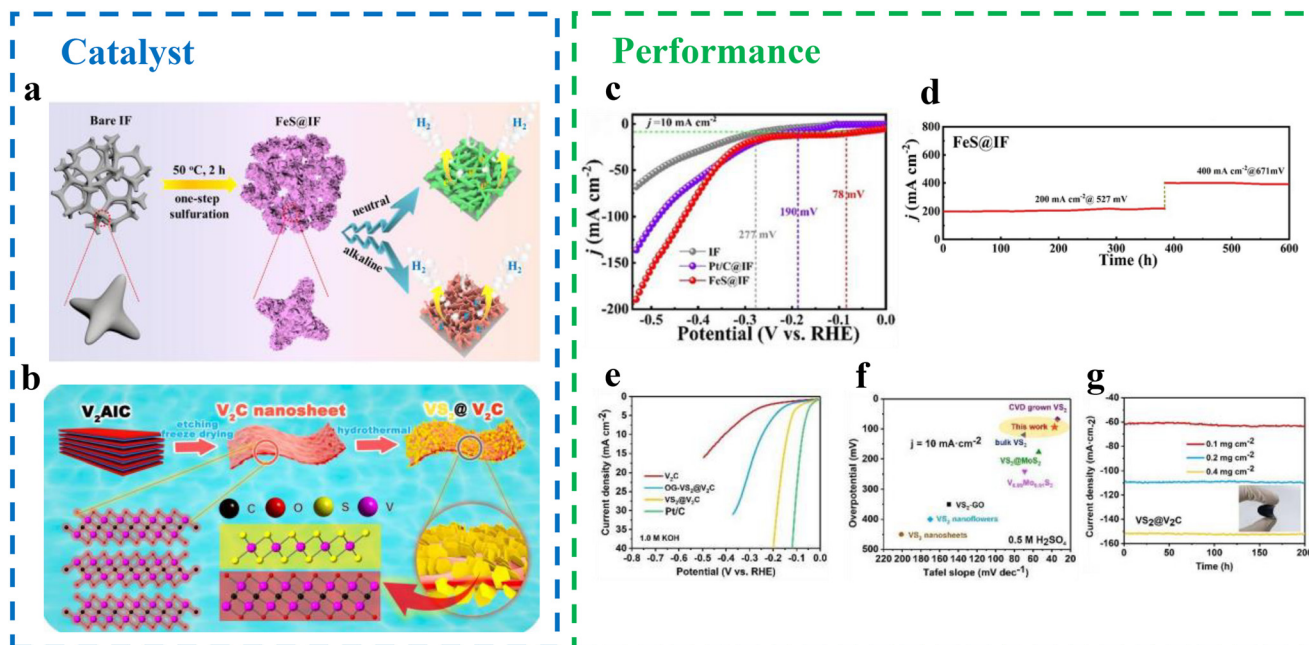
Fig. 13 (a) The optimal synthesis route and condition of FeS@IF electrode.[70] (b) Schematic illustration of the synthesis of $VO_2(p)-V_2C$ host.[71] (c) LSV curves of IF, Pt/C@IF and FeS@IF electrodes in neutral solution.[70] (d) Chronopotentiometric measurement for FeS@IF electrodes in 1.0 M PBS.[70] (e–g) Electrochemical performance of the VS_2-based catalysts.[71]

nificantly enhance HER performance. Wang *et al.*[71] utilized V_2C, a material with high electrical conductivity, as the matrix for composites and developed a novel disulfide water electrolysis strategy based on a three-dimensional structure of vanadium-based two-dimensional carbides. Using a simple one-step hydrothermal method, they successfully synthesized $VS_2@V_2C$ composites (Fig. 13b). Microstructural analysis revealed that vanadium disulfide (VS_2) nanosheets were uniformly embedded vertically within multiple layers of V_2C nanosheets. This architecture prevents aggregation and structural collapse of VS_2 nanosheets, while the wide surface of V_2C serves as a robust support for VS_2. The tight bonding between the two phases facilitates efficient electron transfer, significantly enhancing the HER activity and stability of the composite compared with pure VS_2 or V_2C alone. In minimally treated seawater with varying pH levels, the $VS_2@V_2C$ composites achieved industrial-grade current densities and low overpotentials, comparable to Pt-based catalysts. The biphasic interface enabled rapid charge transfer, while the high specific surface area and unique mosaic structure reduced the free energy of hydrogen ion adsorption. As a result, the composite exhibited excellent electrocatalytic HER performance across different pH environments. In an alkaline electrolyte (pH = 13.8), the composite achieved an overpotential of 164 mV and a Tafel slope of 37.6 mV dec^{-1} at 20 mA cm^{-2} (Fig. 13e). In highly acidic conditions (pH = 0), the overpotential was as low as 138 mV with a Tafel slope of 37.9 mV dec^{-1} (Fig. 13f). At industrial-grade current densities (>100 mA cm^{-2}), $VS_2@V_2C$ outperformed Pt-based catalysts, demonstrating superior efficiency and long-term stability. Additionally, the composite

maintained excellent performance in neutral or mildly alkaline natural seawater, further establishing its potential for practical hydrogen evolution applications (Fig. 13g).

Transition metal oxides. Transition metal oxides have gained significant attention for their efficient HER activity. Du *et al.*[72] developed a robust electrocatalyst composed of $NiFe_2O_4$ deposited on nickel foam for hydrogen production *via* HER-SMOR systems in alkaline water and seawater electrolytes (Fig. 14a). The study also introduced an *in situ* growth strategy for active catalysts on conductive substrates, such as NF, offering potential for large-scale applications. The *in situ* synthesis of $NiFe_2O_4$ on NF ensured high electrical conductivity while enhancing the electrocatalyst's durability under intense electrochemical operating conditions.[73] The dual-electrode system, catalyzed by $NiFe_2O_4/NF||NiFe_2O_4/NF$, achieved industrial-grade current densities (>800 mA cm^{-2}) and a high hydrogen production rate (>1.7 μmol s^{-1} cm^{-2}) at a cell voltage of 2.0 V (Fig. 14c). The system also exhibited excellent faradaic efficiencies, exceeding 96% for HER at the cathode and 95% for formic acid oxidation at the anode (Fig. 14d). Additionally, the $NiFe_2O_4/NF||NiFe_2O_4/NF$ HER-SMOR system demonstrated exceptional stability, maintaining performance after 48 hours of continuous operation (Fig. 14e). Importantly, the system performed well in seawater, exhibiting high activity and stability over 6 hours without significant side reactions, such as the ClOR or OER. Spectroscopic studies and DFT calculations revealed that the SMOR on $NiFe_2O_4/NF$ follows a CO-free pathway, with Fe and Ni sites on the $NiFe_2O_4$ spinel acting as active centers for methanol and OH activation, respectively. This work highlights the potential of $NiFe_2O_4/NF$ for efficient
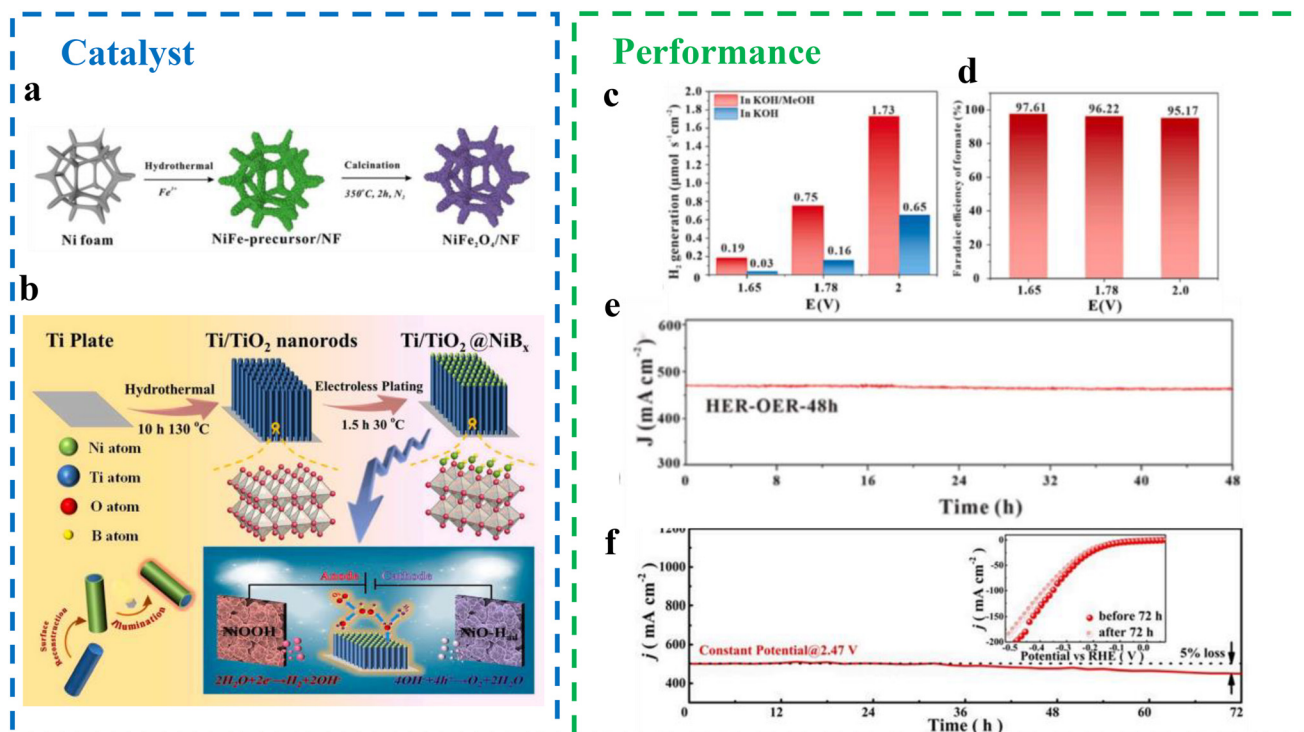
Fig. 14 (a) Schematic illustration of the synthesis of NiFe$_2$O$_4$/NF catalyst.[72] (b) Illustration for preparing the Ti/TiO$_2$@NiB$_x$ electrode toward overall water splitting.[74] (c) Average rates of H$_2$ generation at different potentials.[72] (d) Faradaic efficiencies of formate generation at different potentials in KOH/MeOH.[72] (e) I–t curves of NiFe$_2$O$_4$/NF|NiFe$_2$O$_4$/NF at 1.8 V in 1.0 M KOH for 48 h.[72] (f) Chronopotentiometric measurement of HER at 500 mA cm^{-2}.[74]

and durable hydrogen production in both freshwater and seawater electrolysis systems.

In addition, Fu *et al.*[74] prepared TiO$_2$ nanorods by a one-step alkali etching method and supported them on corrosion-resistant Ti plates (Ti/TiO$_2$). Subsequently, they obtained Ti/TiO$_2$@NiB$_x$ self-supported electrodes by modifying NiB$_x$ catalytic material on Ti/TiO$_2$ by a simple electroless plating method (Fig. 14b). The modification of amorphous NiB$_x$ significantly enhances the catalytic properties of TiO$_2$ semiconductors by increasing the number of reaction sites and improving both photoadsorption and electron–hole pair separation efficiency. The Ti/TiO$_2$@NiB$_x$ electrode demonstrated improved catalytic performance, achieving increases of approximately 16.2% and 14.7% in HER and OER efficiency, respectively, at a current density of 10 mA cm^{-2} under xenon lamp irradiation and electrical energy input. Additionally, the electrode enabled solar-driven overall water splitting at lower potentials and higher current densities. The Ti/TiO$_2$@NiB$_x$ electrode exhibited stable performance during 72 hours of continuous operation, highlighting its potential for industrial applications. Mechanistic analysis revealed that NiO-H$_{ad}$ species were formed at the cathode, while NiOOH active species were detected at the anode, contributing to the enhanced catalytic performance and durability. Moreover, the electrode sustained efficient seawater electrolysis at industrial-grade current densities (500 mA cm^{-2}) during the HER

process for 72 hours, attributed to the high corrosion resistance of the TiO$_2$ layer (Fig. 14f). The modification of amorphous NiB$_x$ increased the specific surface area of the electrode, facilitating the effective separation of electron–hole pairs within the TiO$_2$ semiconductor structure, further improving its efficiency and stability for large-scale seawater electrolysis applications.

Precious metal-based catalysts consistently demonstrate high HER efficiency and stability, benefiting from outstanding catalytic activity and strong corrosion resistance. These findings align with the established role played by platinum group metals in enhancing transition metal alloy catalysts, enabling superior performance even under industrial conditions. Meanwhile, transition metal phosphides and sulfides offer complementary strategies by modulating electronic structures or forming heterostructures to emulate Pt-like activity and achieve high current densities. Additionally, advancements in multiphase interface engineering provide valuable insights into optimizing interfacial electronic properties, paving the way for scalable and durable catalysts. Similarly, studies on oxides and TiO$_2$-based systems highlight alternative approaches, leveraging *in situ* growth techniques and light-assisted catalysis to enhance efficiency and long-term stability in seawater electrolysis. Table 2 summarizes the performance comparison of HER electrocatalysts for seawater electrolysis at industrial-grade current density.

Table 2 Performance comparison of HER electrocatalysts

Catalyst type	Catalyst	Performance	Testing conditions	Ref.
Transition metal phosphides	Cu_3P–FeP@CC	291 mV@500 mA cm^{-2}, 338 mV@1000 mA cm^{-2}/ 406 mV@1000 mA cm^{-2}	1 M KOH/1 M KOH + 0.5 M NaCl	64
Transition metal phosphides	P—Ni_4Mo/CF	551 mV@1000 mA cm^{-2}, 422 mV@500 mA cm^{-2}	1 M KOH + seawater	66
Transition metal phosphides	Co–Ni–P/CP	0.533 V@500 mA cm^{-2}	1 M KOH + 0.5 M NaCl + 0.1 M N_2H_4	67
Transition metal phosphides	Ru/P–$NiMoO_4$@NF	0.173@1000 mA cm^{-2}, 0.242 V@3000 mA cm^{-2}/ 0.299 V@3000 mA cm^{-2}	1 M KOH + 0.5 M NaCl/1 M KOH + seawater	61
Transition metal phosphides	C–Co_2P	192 mV@1000 mA cm^{-2}	1 M KOH, 0.5 M NaCl, 41.2×10^{-3} M $MgCl_2$, 12.5×10^{-3} M $CaCl_2$	75
Transition metal phosphides	1T(0.63)-$MoSe_2$@MoP	358 mV@1000 mA cm^{-2}/ 317 mV@1000 mA cm^{-2}	1 M KOH/1 M KOH + seawater	69
Transition metal sulfide	FeS@IF	490 mV@1000 mA cm^{-2}	1 M KOH + 0.5 M NaCl	70
Transition metal sulfide	VS_2@V_2C	At higher current densities (>100 mA cm^{-2})	In natural seawater (pH = 8.6)	71
Transition metal alloy	NiRh-Cu NA/CF	155 mV@300 mA cm^{-2}	Alkaline simulated seawater electrolyte	62
Transition metal alloy	Pt/Ni–Mo	113 mV@2000 mA cm^{-2}	1.0 m KOH + 0.5 m NaCl	63
Transition metal oxides	$NiFe_2O_4$/NF‖$NiFe_2O_4$/NF	2.0 V@700 mA cm^{-2}	Seawater	72
Transition metal oxides	Ti/TiO_2@NiB_x	72 h@500 mA cm^{-2}	1 M KOH + seawater	74

5.3 Bifunctional electrocatalysts

5.3.1 Nobel metal catalysts.

In industrial-grade seawater electrolysis, bifunctional electrocatalysts incorporating precious metals are widely utilized due to their exceptional performance and stability. Wu et al.[76] developed an amorphous cobalt-based oxide catalyst (Ru–CoO_x/NF) with ultra-low Ru content, capable of efficiently driving water electrolysis at high current densities in both alkaline water and seawater. The catalyst was synthesized through a multi-step process. Initially, a uniformly distributed array of CoZIF-L was grown in situ on nickel foam (CoZIF-L/NF). The precursors were then annealed in air at 250 °C to form CoO_x/NF. Finally, Ru–CoO_x/NF was obtained by doping with Ru followed by calcination (Fig. 15a). The Ru–CoO_x/NF catalyst achieved remarkable HER and OER performance, requiring overpotentials of only 252 mV and 370 mV, respectively, to reach a current density of 1000 mA cm^{-2} in 1 M KOH (Fig. 15c–e). These values outperform commercial Pt/C and RuO_2 catalysts. The catalyst also demonstrated excellent electrochemical stability under high current densities. When used in an electrolyzer device, Ru–CoO_x/NF required ultra-low voltages of 2.2 V in alkaline water and 2.62 V in seawater to sustain a current density of 1000 mA cm^{-2}, surpassing the performance of commercial Pt/C and RuO_2 catalysts (Fig. 15f). These results highlight the Ru–CoO_x/NF catalyst's potential for efficient and cost-effective seawater electrolysis under industrial conditions. Similarly, Kong et al.[77] developed Ru–Co_3O_4 by embedding Ru within the thin layers of Co_3O_4, creating a dense Ru–O–Co cluster designed for decomposing water and seawater (Fig. 15b). This novel structure speeds up the generation and breakdown of active H_2O in the hydrogen evolution process, ensuring rapid evolution of essential "active oxygen" components in the OER. As a result, it enhances the intrinsic kinetics of the HER/OER. The components integrated in these thin-layer structures effectively promote the formation and utilization of active sites. In comparison with Ru/C and Pt/C, Ru–Co_3O_4 exhibits superior HER mass activity, cost-effectiveness, and OER mass activity. When utilized in AEM electrolyzers, Ru–Co_3O_4 delivers an industrial-grade current density, splitting water at 2.03 V for 1 A cm^{-2} and seawater at 1.71 V for 0.41 A cm^{-2}, both over twice that of Ru/C and Pt/C components (Fig. 15g–i), significantly improving the efficiency of water and seawater decomposition.

5.3.2 Transition metal compounds

Transition metal oxide. Transition metal oxides exhibit high intrinsic electrocatalytic activity for the HER and OER due to their strong hydrolysis ability across various valence states. A hybrid strategy has been proposed to enhance the water electrolysis performance of transition metal oxide electrodes. Guo et al.[78] introduced Lewis acid layers (e.g., Cr_2O_3) onto the surface of commonly used catalysts to regulate the local reaction microenvironment, enabling direct seawater electrolysis without the need for alkalization or acidification (Fig. 16a). This approach is versatile and can be applied to various catalysts without redesigning the catalyst or the electrolytic cell. Using Lewis acid-modified electrodes (Cr_2O_3–CoO_x), the system achieved stable performance for over 100 hours in real seawater, splitting water molecules and capturing large amounts of hydroxyl anions (OH^-) generated in situ around the catalyst. The performance was comparable to that of pure water PEM electrolytic cells, delivering a current density of 1.0 A cm^{-2} at 60 °C with a cell voltage of 1.87 V in a flowing seawater electrolyzer (Fig. 16b). The effectiveness of this strategy lies in the preferential enrichment of OH^- ions around the catalyst, which suppresses chlorination reactions and minimizes carbon contamination on the catalyst surface. Additionally, the strong binding between OH^- and the Lewis acid layer reduces the formation of precipitates, preventing the
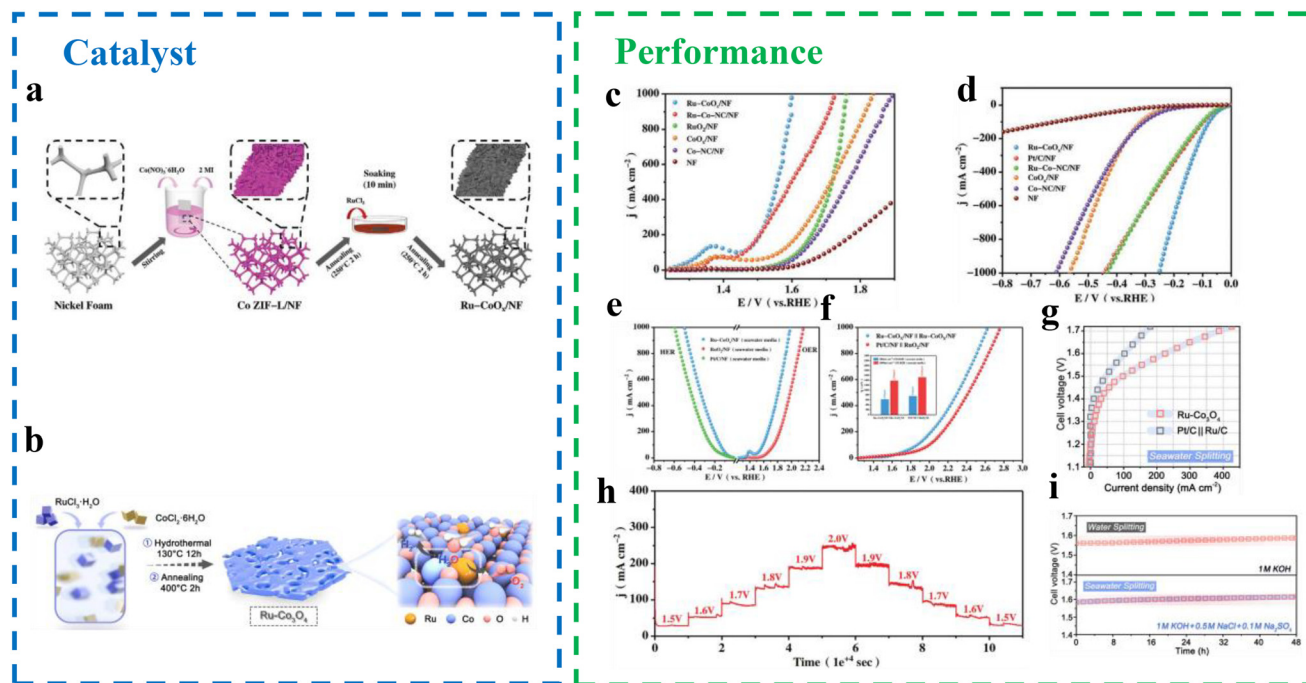
Fig. 15 Schematic illustration of the fabrication of (a) Ru–CoO$_x$/NF.[76] (b) Ru–Co$_3$O$_4$ with Ru–O–Co moieties.[77] (c–e) OER and HER polarization curves of Ru–CoO$_x$/NF and other samples in 1 M KOH and seawater media.[76] (f) Polarization curves of Ru–CoO$_x$/NF||Ru–CoO$_x$/NF and Pt/C/NF|| RuO$_2$/NF toward overall water splitting in seawater media.[76] (g) Seawater splitting performance of Ru–Co$_3$O$_4$ and Pt/C||C.[77] (h) Multipotential stability test chart of Ru–CoO$_x$/NF in seawater media.[77] (i) Chronopotentiometry stability tests of Ru–Co$_3$O$_4$ for water and seawater electrolysis.[77]
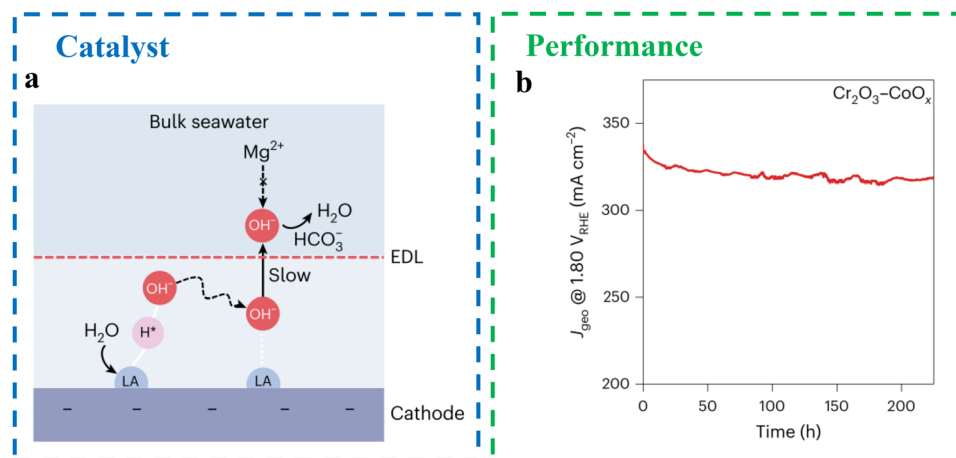

Fig. 16 (a) Schematic diagram of local alkaline microenvironment generation on Lewis acid-modified cathode. LA = Lewis acid. EDL = electrical double layer.[78] (b) Long-term durability test of 6 at% Cr$_2$O$_3$–CoO$_x$ anode at 1.80 V RHE in natural seawater.[78]

capture of OH$^-$ by Mg^{2+} and Ca^{2+} cations in seawater electrolytes. The properties of the anode and cathode in Lewis acid-modified flowing seawater electrolytic cells are comparable to those of pure water catalysts using RuO$_2$ and Pt/C, respectively, demonstrating a promising pathway for efficient and sustainable seawater electrolysis.

Transition metal phosphide. Transition metal phosphide catalysts often excel in either the OER or HER but rarely achieve high activity in both simultaneously. Chang *et al.*[79] addressed

this limitation by synthesizing Fe and phosphorus double-doped nickel selenide nanoporous films (Fe, P-NiSe$_2$ NFs) as a bifunctional catalyst with excellent performance for seawater electrolysis (Fig. 17a). The enhanced performance of Fe, P-NiSe$_2$ NFs is attributed to the weaker bond strength of Se–H compared with P–H, which allows selenides to trap reactants more effectively and accelerate the discharge step. Simultaneously, the slightly oxidized surface of Fe, P–NiSe$_2$ prevents dissolution, ensuring stable and balanced HER and
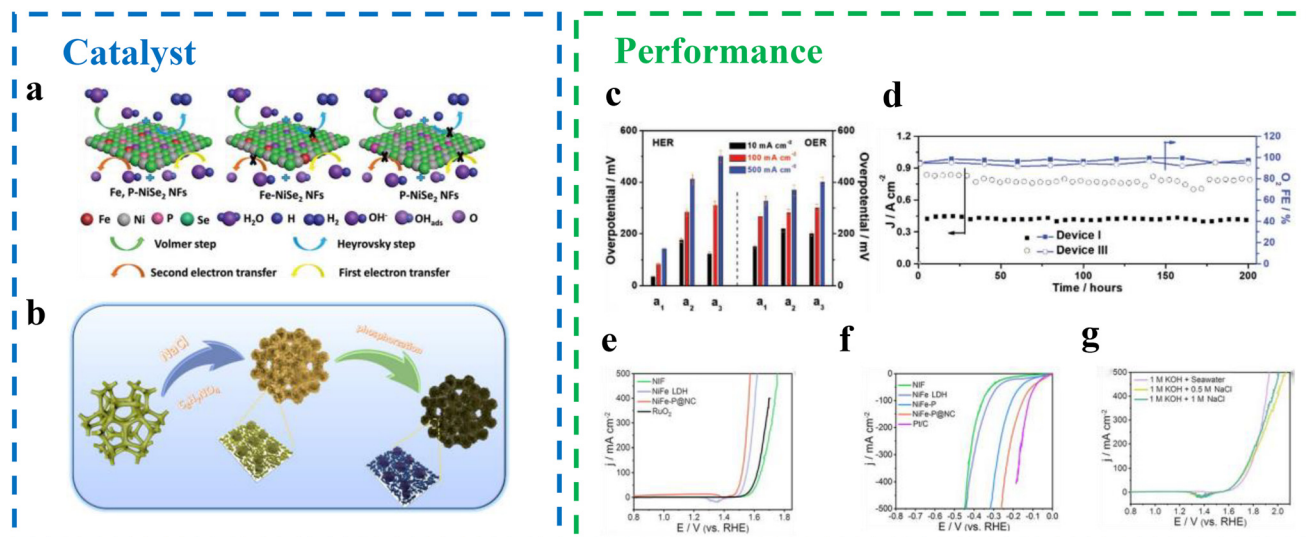
Fig. 17 (a) Illustration for the HER and OER reaction pathways on different catalysts.[79] (b) The synthesis of NiFe-P@NC.[80] (c) Comparison of the HER and OER overpotentials required to achieve current densities of 10, 100, and 500 mA cm^{-2} for: (a$_1$) Fe,P-NiSe$_2$ NFs, (a$_2$) Fe–NiSe$_2$ NFs, and (a$_3$) P–NiSe$_2$ NFs.[79] (d) Long-term operation test of asymmetric device I and symmetric device III seawater electrolyzers showing high selectivity and stability at 1.8 V.[79] Plots of the (e) OER and (f) HER performance tests of the prepared electrocatalytic materials in KOH (1 M) + seawater.[80] (g) Plots of NiFe-P@NC||NiFe-P@NC in alkaline seawater for overall water-splitting.[80]

OER activities. Experimental results and DFT calculations revealed that Fe-dopants act as active HER sites, while Ni atoms adjacent to Fe-dopants serve as catalytic centers for the OER. The dual role played by P-dopants was identified as improving electrical conductivity and forming a surface passivation layer containing P–O species, which prevents selenium dissolution and enhances catalyst stability. Self-supported on carbon cloth, Fe, P–NiSe$_2$ NFs can be directly utilized as bifunctional catalysts in seawater electrolyzers without the need for additional additives. The catalyst achieved an industrial-grade current density of 0.8 A cm^{-2} at 1.8 V, maintaining performance for over 200 hours with an oxygen faradaic efficiency exceeding 92% (Fig. 17c and d). In summary, Fe doping enhances OER selectivity and faradaic efficiency, while P doping improves electrical conductivity and stabilizes the catalyst by forming a protective P–O passivation layer. These features make Fe, P-NiSe$_2$ NFs a promising candidate for efficient and durable bifunctional catalysis in seawater electrolysis.

Phosphorylation is a widely used technique to enhance the water electrolysis activity of transition metal catalysts. To further boost catalytic performance, Chen et al.[80] developed a novel nitrogen-doped carbon-coated nickel phosphide catalyst (NiFe-P@NC) using corrosion engineering combined with a mild phosphorylation process (Fig. 17b). NiFe-P@NC is a bifunctional catalyst with exceptional electrocatalytic performance, enabling efficient HER and OER reactions. The unique hierarchical morphology of NiFe-P@NC, characterized by an interconnected three-dimensional porous structure, significantly increases the catalyst's surface area and the number of active sites, thereby enhancing its catalytic efficiency. Experimental results demonstrated that NiFe-P@NC achieved

excellent HER activity, requiring an overpotential of only 40 mV to reach 10 mA cm^{-2}. Additionally, its robust structure and chemical composition conferred remarkable stability in 1 M KOH, resisting corrosion and oxidation during prolonged operation. In a 1 M KOH + seawater electrolyte, NiFe-P@NC displayed outstanding bifunctional catalytic performance, with low overpotentials of 149 mV and 280 mV required for the HER and OER at 100 mA cm^{-2}, respectively (Fig. 17e and f). When assembled into a NiFe-P@NC||NiFe-P@NC electrolyzer, the system achieved current densities of 100 mA cm^{-2} and 500 mA cm^{-2} at low voltages of 1.77 V and 1.93 V, respectively, in alkaline seawater electrolytes (Fig. 17g). This superior performance is attributed to NiFe-P@NC's high electrical conductivity and catalytic activity. Overall, NiFe-P@NC demonstrates exceptional potential as a bifunctional electrocatalyst for efficient HER and OER reactions, making it a promising candidate for industrial water electrolysis applications.

Transition metal sulfide. Transition metal sulfides are well known for their bifunctional HER and OER electrocatalytic activity. To optimize their performance for high-current-density seawater electrolysis, novel multifaceted electrocatalysts combining sulfur with other transition metals have been developed. Yao et al.[81] synthesized a hybrid electrode with local photothermal effects (LPTE) via a one-pot hydrothermal reaction. This electrode features a ternary nickel–bismuth–sulfur nanosheet array supported on nickel foam (Ni$_3$Bi$_2$S$_2$@NF) (Fig. 18a). The Ni$_3$Bi$_2$S$_2$@NF electrode demonstrated significant efficiency improvements in HER and OER, with increases of 44% and 35%, respectively, under light irradiation. These enhancements are attributed to the intrinsic LPTE effect of bismuth, the strong phase stability of Ni$_3$Bi$_2$S$_2$, and the synergistic properties of its layered structure. The
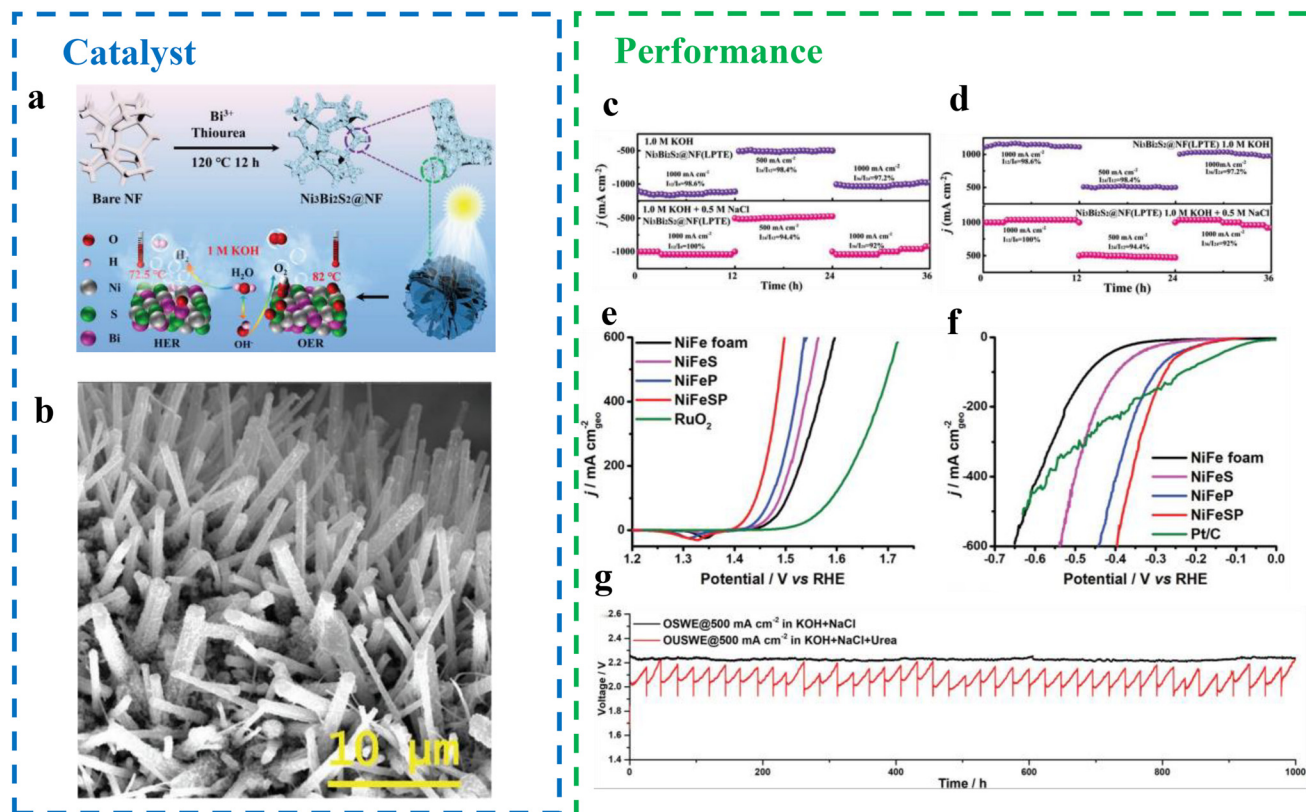
Fig. 18 (a) Illustration for preparing $Ni_3Bi_2S_2$@NF electrode and overall water splitting process with LPTE.[81] (b) SEM image of NiFeSP-NT.[82] (c) Long-time LPTE stability experiments of $Ni_3Bi_2S_2$@NF under different lager current density each for 12 h in 1.0 M KOH and simulated seawater solutions.[81] (d) Chronopotentiometric measurements of long-term stability of coupled-LPTE $Ni_3Bi_2S_2$@NF under different constant voltages for 12 h in 1.0 M KOH and 1.0 M KOH + 0.5 M NaCl solution.[81] Polarization curves of the (e) OER and (f) HER, recorded at 5 mV s^{-1}.[82] (g) Chronopotentiometric curves of the OSUWE and OSWE for the NiFeSP electrode pair.[82]

LPTE effect facilitates localized heat generation, accelerating reaction kinetics and improving catalytic efficiency. The electrode also showed exceptional performance in a full water-splitting system. Using $Ni_3Bi_2S_2$@NF as a bifunctional electrode, the system achieved an ultra-low voltage of 1.40 V under LPTE at a current density of 10 mA cm^{-2}. It maintained stability for over 36 hours at industrial-grade current densities of 500–1000 mA cm^{-2} (Fig. 18c and d). These results highlight the superior electrocatalytic activity and durability of $Ni_3Bi_2S_2$@NF, making it a promising candidate for seawater electrolysis and other water-splitting applications. The integration of LPTE with a robust ternary structure underscores its potential for energy-efficient and stable hydrogen production technologies.

The sulfidation process is a powerful method to enhance the water electrolysis performance of transition metal phosphides. Yu et al.[82] developed a self-supported nickel–iron phosphide (NiFeSP) nanotube array electrode using a two-step sulfidation/phosphorylation process. These NiFeSP nanotubes feature abundant NiFeS/NiFeP heterogeneous surfaces and low-coordination metal sites, which synergistically improve electrocatalytic activity (Fig. 18b). Experimental results showed that the NiFeSP electrode demonstrated outstanding HER and

OER performance in simulated alkaline aqueous solutions (KOH + NaCl), achieving overpotentials of 380 mV for HER and 260 mV for OER at 500 mA cm^{-2} (Fig. 18e and f). The electrode exhibited excellent durability, maintaining performance for 1000 hours. To address the issue of competing chloride evolution reactions during the OER, the researchers replaced the OER with a urea oxidation reaction (UOR), significantly reducing the overall energy input required for alkaline water electrolysis. Theoretical calculations confirmed that the heterogeneous interfaces and low-coordination metal sites in the NiFeSP nanotubes lower the energy barrier for the rate-determining step, enhancing catalytic performance. Additionally, the electrode demonstrated excellent activity for the UOR, enabling efficient urea-mediated alkaline water electrolysis. By coupling the UOR with the HER, the bifunctional NiFeSP electrode effectively catalyzed alkaline water electrolysis at 500 mA cm^{-2} for 1000 hours at a voltage of 1.938 V without significant performance degradation (Fig. 18g). These findings highlight the potential of NiFeSP nanotube arrays for energy-efficient and durable water electrolysis systems.

Transition metal nitride. Transition metal nitrides are known for their strong electrocatalytic activity in the HER and OER, along with high corrosion resistance, electrical conductivity,

and mechanical strength. However, their electrocatalytic performance in high-current-density seawater electrolysis still has room for improvement. One effective strategy to enhance bifunctional catalytic activity is designing heterogeneous structures with coupled functionalized species. Ning *et al.*[83] successfully synthesized Fe-doped Ni&$Ni_{0.2}Mo_{0.8}$N on nickel foam as a highly active HER catalyst (Fig. 19a). The study revealed that varying Fe doping concentrations significantly influenced the rod size of the catalyst and its intrinsic HER activity. During rapid electrochemical reconstruction, Fe_x–Ni&$Ni_{0.2}Mo_{0.8}$N transformed *in situ* into Fe_x&Mo–NiO, exhibiting strong OER activity. Tafel analysis and DFT calculations indicated that Fe doping shifted the rate- and potential-determining steps of Mo–NiO in the OER mechanism, enhancing catalytic efficiency. Systematic investigations compared the performance of $Fe_{0.01}$–Ni&$Ni_{0.2}Mo_{0.8}$N‖$Fe_{0.01}$&Mo–NiO electrolyzers in alkaline freshwater and seawater electrolysis. The electrolyzer achieved nearly 100% Faraday efficiency for H_2 and O_2 production in both 1 M KOH and seawater. It demonstrated record-high current densities in seawater electrolysis at 1.7 V (Fig. 19c). Under quasi-industrial conditions, its performance in highly alkaline seawater (60 °C) closely matched that in alkaline freshwater, with only a 1.49% voltage increase at 1000 mA cm^{-2}. The electrolyzer maintained excellent stability in 6 M KOH and seawater at 60 °C, even under harsh conditions (Fig. 19d). These findings underscore the potential of Fe-doped Ni&$Ni_{0.2}Mo_{0.8}$N with Mo–NiO heterostructures as bifunctional catalysts for efficient and durable H_2 and O_2 production in seawater electrolysis. Moreover, the study highlighted the importance of addressing redox substance concentrations in alkaline seawater, which are higher than in alkaline freshwater and may lead to catalyst poisoning and corrosion. Careful electrolyte selection and treatment are crucial to ensur-

ing the long-term stability and efficiency of catalysts in seawater electrolysis systems.

Metal–N bonding in transition metal nitrides induces a contraction of the metal d-band, raising the energy levels near the Fermi energy. This redistribution of the density of states enhances the electrocatalytic activity of transition metal nitrides, making them comparable to noble metals. Wang *et al.*[84] reported an advanced electrocatalyst, NiFeLDH@Ni_3N/NF, which features a three-dimensional hierarchical heterogeneous structure composed of nano/micro sheets fabricated on nickel foam (Fig. 19b). This electrocatalyst demonstrates excellent electrical conductivity, abundant active sites, and a synergistic interaction between NiFeLDH and Ni_3N, leading to superior electrocatalytic performance and stability. When used as a bifunctional electrocatalyst, NiFeLDH@Ni_3N/NF enabled a two-electrode electrolyzer to achieve a current density of 500 mA cm^{-2} at a cell voltage of 1.80 V with remarkable stability over 100 hours (Fig. 19e). Additionally, Wang *et al.* developed a carbon-coated nickel nitride micro sheet array (Ni_xFe_yN@C/NF) as a highly efficient and durable electrocatalyst for seawater electrolysis. This electrocatalyst leverages a porous micro sheet array structure to provide abundant active sites, ideal surface superhydrophobicity and superhydrophilicity to enhance mass transfer, and efficient charge transport due to the synergistic coupling between Ni_xFe_yN and the carbon coating. The durability of Ni_xFe_yN@C/NF is attributed to the protective effect of the carbon coating and the intrinsic corrosion resistance of Ni_xFe_yN. These features make it a promising electrocatalyst for long-term and high-performance seawater electrolysis, offering a practical solution for sustainable hydrogen production in harsh environments.

5.3.3 Non-metallic catalyst. Non-metallic catalysts have gained significant attention for seawater electrolysis due to
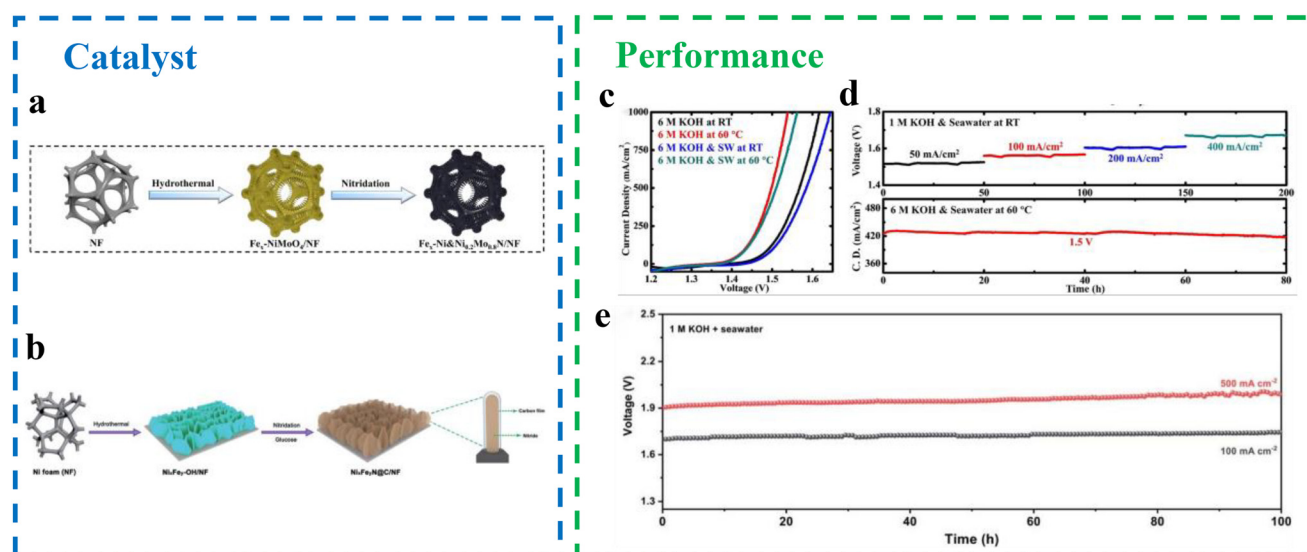

Fig. 19 (a) Synthesis process for Fe_x–Ni&$Ni_{0.2}Mo_{0.8}$N microrods on NF.[83] (b) The preparation procedure of Ni_xFe_yN@C/NF.[84] (c) Alkaline water and seawater electrolysis performance of $Fe_{0.01}$–Ni&$Ni_{0.2}Mo_{0.8}N_8Fe_{0.01}$&Mo–NiO under industrial conditions.[83] (d) Stability measurements during alkaline seawater electrolysis.[83] (e) Durability tests of this electrolyzer at 100 and 500 mA cm^{-2}.[84]
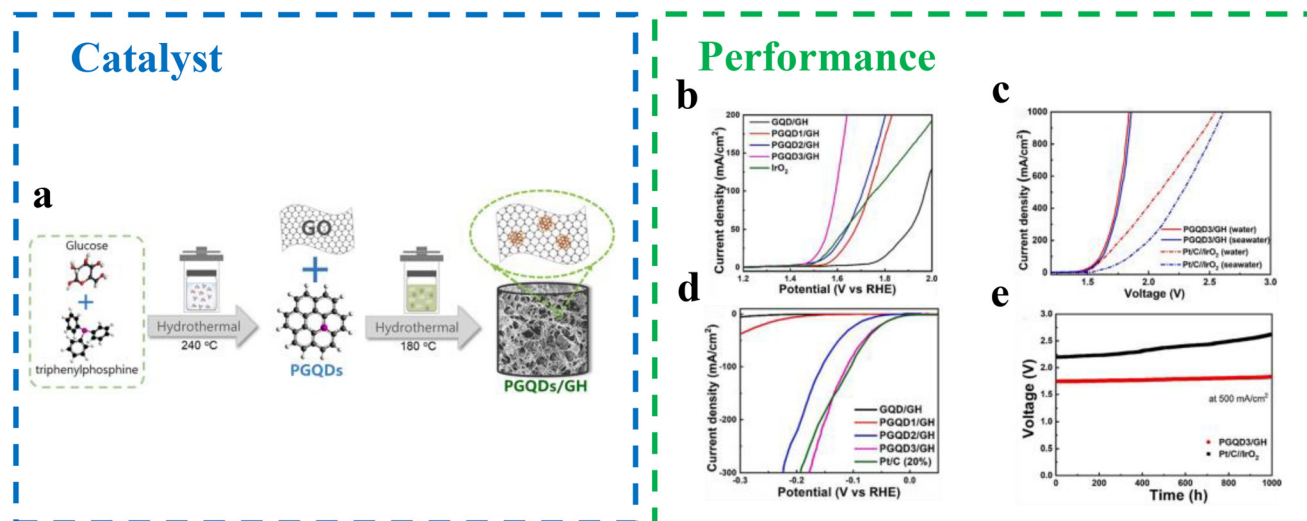
Fig. 20 (a) Schematic illustration for the preparation of PGQDs and PGQDs/GH composites. (b) OER and (d) HER LSV curves of the prepared PGQDs/GH composites and commercial IrO_2 in 1 M KOH. (c) LSV curves for electrolysis of fresh water and seawater using PGQDs/GH electrodes as both anode and cathode. (e) Chronopotentiometric stability of seawater electrolysis using PGQDs/GH electrodes for 1000 h.[85]

their distinctive chemical properties. Tam *et al.*[85] developed an innovative material, PGQDs/GH, composed of p-doped graphene quantum dots and reduced graphene hydrogel (Fig. 20a). This material was investigated for its catalytic role in hydrogen production from seawater *via* electrolysis. PGQDs/GH features a high surface area, a porous structure, and multiple conductive pathways facilitated by diverse p-configurations, including PC_3, PO_4, and PO_3 bonds. The PC_3 bond plays a critical role in modulating the bandgap and creating active sites for both photocatalytic and electrocatalytic seawater splitting reactions, improving charge transfer and reaction kinetics. In an alkaline environment, PGQDs/GH demonstrated impressive performance with a low overpotential of

178 mV for the HER at 300 mA cm^{-2} (Fig. 20d) and 289 mV for the OER at 20 mA cm^{-2} (Fig. 20b). These results surpassed the performance of traditional Pt/C and IrO_2 catalysts. In seawater electrolysis, the PGQDs/GH electrode exhibited a low cell voltage of 1.62 V at 100 mA cm^{-2}, outperforming commercial Pt/C//IrO_2 electrodes (Fig. 20c). Furthermore, the catalyst demonstrated better durability, maintaining stable performance over 1000 hours of continuous operation at 500 mA cm^{-2} with only a 5% voltage drop (Fig. 20e). These results underscore the potential of PGQDs/GH as an efficient and durable non-metallic catalyst for seawater electrolysis, offering a cost-effective and sustainable alternative for hydrogen production.

Table 3 Performance comparison of bifunctional electrocatalysts

Catalyst type	Catalyst	Performance	Testing conditions	Ref.
Transition metal oxides	Ru–CoO_x/NF||Ru–CoO_x/NF	970 mV@1000 mA cm^{-2}/ 1390 mV@1000 mA cm^{-2}	1 M KOH prepared with deionized water and potassium hydroxide/1 M KOH prepared with seawater and potassium hydroxide	76
Transition metal oxides	Cr_2O_3–CoO_x||Cr_2O_3–CoO$_x$	1.87 V@1000 mA cm^{-2}	In natural seawater, 60 °C	78
Transition metal phosphides	Fe, P-$NiSe_2$ NFs||Fe, P-$NiSe_2$ NFs	1.8 V@0.8 A cm^{-2}/ (141 mV-HER, 317 mV-OER) @500 mA cm^{-2}	Natural seawater/1 M KOH	79
Transition metal phosphides	NiFe-P@NC||NiFe-P@NC	217 mV-HER@1000 mA cm^{-2}/ 340 mV-OER@500 mA cm^{-2}, 1.93 V@500 mA cm^{-2}	1 M KOH/Alkaline seawater electrolyte	80
Transition metal sulfide	$Ni_3Bi_2S_2$@NF||$Ni_3Bi_2S_2$@NF	2.34 V@500 mA cm^{-2}, 3.10 V@1000 mA cm^{-2}	1.0 M KOH + 0.5 m NaCl	81
Transition metal sulfide	NiFeSP||NiFeSP	(380 mV-HER, 260 mV-OER) @500 mA cm^{-2}, 2.225 V@500 mA cm^{-2}	1.0 M KOH + 0.5 m NaCl	82
Transition metal nitride	$Fe_{0.01}$-Ni&$Ni_{0.2}Mo_{0.8}$N|| $Fe_{0.01}$&Mo-NiO	1.585V@500 mA cm^{-2}, 1.644 V@1000 mA cm^{-2}	6 M KOH + seawater	83
Transition metal nitride	Ni_3FeN@C/NF(anode)|| Ni_3N@C/NF(cathode) (Ni_xFe_yN@C/NF)	1.91V@500 mA cm^{-2}	1 M KOH + seawater	84

Noble metal-based catalysts demonstrate outstanding catalytic efficiency and stability, outperforming traditional Pt/C and RuO$_2$ due to innovative structural enhancements and precise elemental doping strategies. In parallel, transition metal compounds leverage unique doping and surface engineering techniques to balance HER and OER activities, achieving excellent durability and industrial-level current densities. Contrastingly, approaches incorporating non-metallic catalysts provide a sustainable, cost-effective alternative with competitive performance metrics and long-term stability, emphasizing their potential for scalable hydrogen production. Furthermore, novel strategies, including photothermal effects and heterogeneous structures in transition metal sulfides and nitrides, showcase synergistic interactions and advanced reaction kinetics, pushing efficiency boundaries. While these studies collectively underline the versatility and potential of bifunctional catalysts, they also reveal differing optimization paths, highlighting the need for targeted designs tailored to specific operational conditions and sustainability goals. Table 3 summarizes the performance comparison of bifunctional electrocatalysts for seawater electrolysis at industrial-grade current density.

6 Device designs and techno-economic analysis for direct seawater electrolysis

Designing direct seawater electrolysis (DSE) devices can pose greater challenges than catalyst design. Effective DSE operation necessitates innovative catalyst engineering and optimizing the catalyst layer characteristics. It also entails careful planning for components like porous transport layers, membranes, and operational parameters. Considering the unique requirements of different DSE setups for catalyst performance, this section will systematically examine the application and techno-econ-omic analysis (TEA) of DSE devices to enhance their efficiency.[86]

6.1 Electrodes

Direct seawater electrolysis encompasses two distinct electrode preparation methods: the traditional method involves the use of ion membranes with catalyst-coated substrates and membranes, whereas the alternative method utilizes ion membrane-free self-supporting electrodes.

6.1.1 Traditional method using ionic polymers. The catalyst-coated substrate (CCS) and catalyst-coated membrane (CCM) methods are two key approaches for manufacturing membrane electrode assemblies in anion exchange membrane water electrolyzers (AEMWE). Each method has distinct advantages and limitations (Fig. 21a). Park *et al.*[87] studied the CCS and CCM methods used to manufacture MEA to find the best method from the perspective of AEMWE performance. In the CCS method, the catalyst layer (CL) is deposited on the liquid/gas diffusion layer (LGDL). While this approach is straightforward and suitable for large-scale production, it often results in insufficient contact between the CL and the membrane, leading to higher contact resistance and reduced performance. Without hot pressing, MEAs produced using the CCS method exhibit significantly lower current densities, making AEMWE operation unreliable due to the high contact resistance. However, the performance improves when hot pressing is applied, with an optimal temperature of 70 °C determined to enhance CL/membrane contact without chemically damaging the membrane. Exceeding this temperature can cause membrane degradation and render AEMWE inoperable. In comparison, the CCM method directly applies the CL to the membrane, achieving superior CL/membrane interface contact. MEAs produced using the CCM method demonstrate the best performance, with lower ohmic resistance, charge transfer resistance, and mass transfer resistance compared with those manufactured using the CCS method, even with hot pressing. For MEAs fabricated with the hot pressing CCS method, the
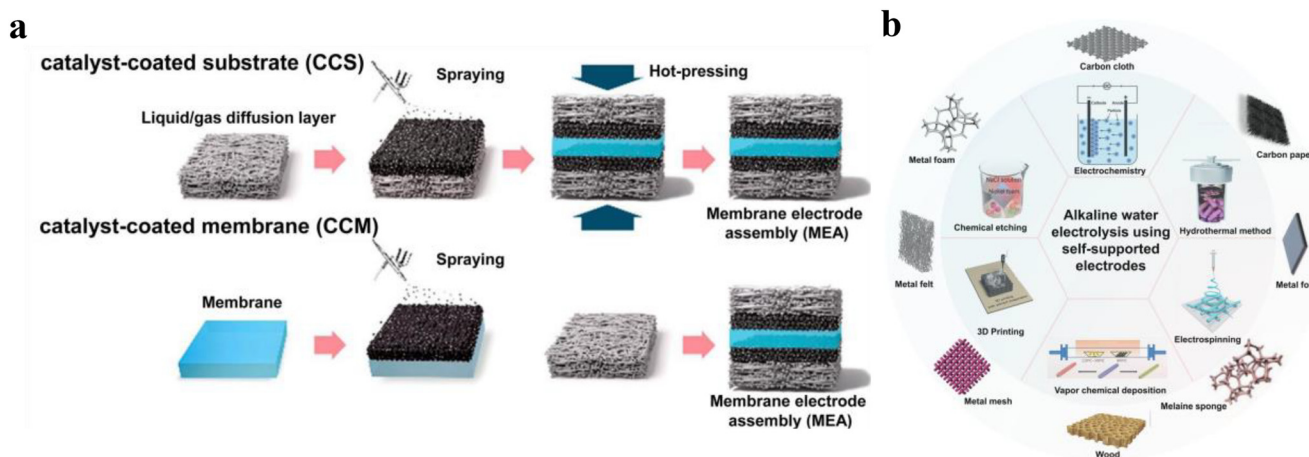

Fig. 21 Schematic of conventional MEA preparation methods with ionomers: (a) CCS and CCM methods.[87] (b) Summary of fabrication methods and used substrates of self-supported electrodes.[88]

total thickness, including the gas diffusion layer (GDL), is approximately 540 µm, compared with 810 µm for MEAs produced using the CCM method. This compression of the GDL during the CCS process alters the pore structure, leading to increased mass transfer resistance, particularly evident in low-frequency impedance arcs. In contrast, the CCM method maintains the GDL structure, minimizing mass transfer resistance and improving overall performance. In conclusion, while hot pressing improves the CCS method's performance, the CCM method is superior due to its lower resistances and better overall efficiency. As such, the CCM method is considered the optimal approach for fabricating high-performance MEAs for AEMWE systems.

6.1.2 Self-supporting electrode without ionomer. Support electrodes are constructed using active materials that naturally integrate with the liquid/gas diffusion layer, including carbon cloth, carbon paper, metal foil, melamine sponge, wood, metal mesh, metal felt, and metal foam. Unlike powder catalysts, self-supporting electrodes eliminate the need for polymer adhesives, thereby maximizing the exposure of active sites and facilitating efficient bubble diffusion. Enhancements in mechanical stability have been achieved, ensuring robust electron transfer between active materials and substrates. Methods for fabricating self-supporting catalysts currently include electrochemical approaches, hydrothermal procedures, electrospinning techniques, gas-phase chemical deposition processes, 3D printing methodologies, and chemical etching methodologies (Fig. 21b).[88]

6.2 Membranes

Hydroxide ion-conducting membranes are critical components in alkaline direct seawater electrolysis, enabling hydroxide ion transport, establishing internal pathways, and preventing hydrogen gas infiltration to mitigate potential hazards. These membranes must possess key physical and chemical properties, including high ion conductivity, alkaline stability, mechanical strength, low gas permeability, an appropriate expansion ratio, and cost-effectiveness. In practical applications, hydroxide ion-conducting membranes for alkaline water electrolysis need to combine superior hydroxide ion conductivity, robust alkaline stability, mechanical durability, minimal hydrogen gas permeation, and affordability to support large-scale industrial use. Currently, commercial

membranes for alkaline water electrolysis are classified into three categories: porous membranes (PMs), ion dissolution membranes (ISMs), and anion exchange membranes (AEMs) (Fig. 22). Porous membranes are typically composed of chemically inert polymer substrates and hydrophilic inorganic fillers, offering excellent resistance to alkalis. However, challenges such as high ohmic resistance and inadequate gas barrier properties in porous membranes have driven researchers to develop ion dissolution membranes tailored for alkaline liquid water electrolysis. Anion exchange membranes feature a polymer framework with nitrogen-containing cations that enable hydroxide ion transport through cationic sites within the membrane. This structure makes AEMs highly desirable due to their excellent hydroxide ion conductivity. A wide range of commercial anion exchange membranes is currently employed in water electrolysis, underscoring their significance in achieving efficient and reliable electrolysis processes.[88]

6.3 Direct seawater electrolysis system

AEMWE represents an innovative approach for water electrolysis, merging the strengths of AWE and PEMWE. It employs affordable non-precious metal catalysts to enhance energy efficiency. AEM plays a crucial role in halting cation diffusion from the anode to the cathode. Moreover, operating under high pH levels minimizes the competitive CER for OER, making AEMWE a more appealing option compared with PEMWE and generating significant enthusiasm for DSE.[86] In practical applications, AEMWE continues to encounter issues like underutilization of catalysts, restricted mass transport, and elevated ohmic resistance. To mitigate polarization losses across kinetic, mass transport, and ohmic domains, the overall design of MEA needs to prioritize expanding three-phase boundaries and mass transport pathways within the CL, as well as enhancing the membrane/CL and CL/LGDL interfaces. Nanostructured MEAs with organized three-phase transport channels in the CL offer advantages over traditional MEAs, enhancing electrolysis performance and longevity. Strategies to boost hydroxide conductivity include increasing ion exchange capacity (IEC) and establishing ion conduits within the membrane. Transforming the CL/membrane interface structure from two-dimensional to three-dimensional can improve ion transport and stability at the interface. Additionally, optimizing MPL with suitable porosity, pore size, and thickness aids
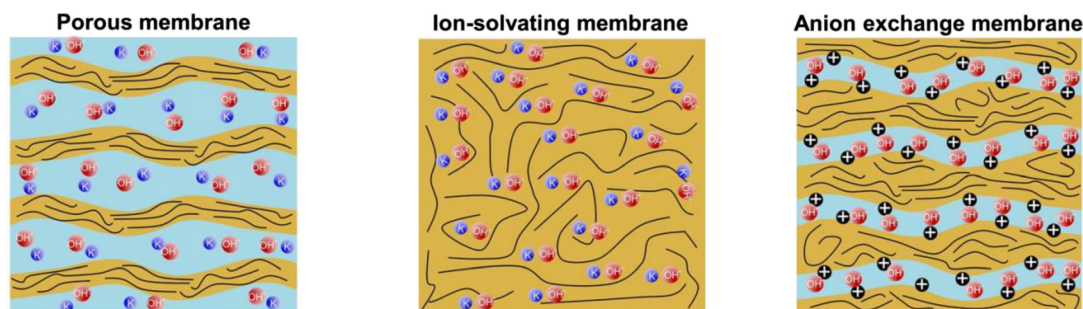

Fig. 22 Three approaches for hydroxide conducting membranes, including PMs, ISMs, and AEMs.[88]

in creating advanced CL/LGDL interfaces, leading to improved catalyst utilization efficiency and reduced mass transport losses.[88]

6.4 Techno-economic analysis

TEA offers a method for evaluating the financial effects of industrial procedures, goods, or services. The integration of TEA and process modeling is vital for assessing the practicality of introducing new technologies and advocating for their possible enhancements and market uses. In particular, examining techno-economic elements is crucial for recognizing how new technologies are integrated into creative business frameworks and acts as a fundamental motivator for important research and development paths. The net present value (NPV) and levelized cost of hydrogen (LCOH) stand as two commonly utilized measures to demonstrate the outcomes of TEA analysis. Conventional NPV analysis employs the discount rate to gauge the true worth of upcoming investments, acting as an effective instrument for evaluating the profit prospects of investment endeavors, depending on their positive or negative correlation with NPV. Nonetheless, the considerable impact of selecting discount rates, capital expenditures, calculating cash flows, and investment magnitude on NPV makes its application in contrasting various hydrogen production systems difficult. Consequently, LCOH, serving as a benchmark for evaluating the cost-effectiveness of hydrogen production methods, has undergone extensive analysis and comparison.[89]

DSE encounters difficulties in terms of both operational and capital expenditures. Table 4 displays a comparative analysis of the specific energy consumption (SEC) in seawater desalination *versus* traditional polymer electrolysis membrane water electrolysis. Only a mere 0.05–0.14% of the total SEC in H_2 production is derived from the process of seawater desalination. Given the substantial role played by electricity in the economic and ecological facets of H_2 generation, a comparable pattern in its influence on operational expenses and the LCOH is expected. Additionally, the latest studies show that the initial expenses for DSE (surpassing 6000 $ per kW) exceed double those of conventional electrolysis cell capital, ranging from 920–1725 USD per kW for alkaline water electrolysis and 1610–2668 USD per kW for PEMWE. The impact of seawater makeup, along with chlorine's cross-effects and corrosion, necessitates significant financial outlay for DSE, resulting in the regular substitution of membranes, catalysts, and system parts.

Utilizing pre-fading methods for the indirect breakdown of seawater allows us to avoid possible adverse effects and risks of corrosion. Yet, this method demands greater energy investment,

diminishing its economic appeal. Moreover, the large size of an independent seawater desalination system restricts the flexibility of the seawater electrolysis system. Consequently, an analysis of processes and economics leads to the primary inference that DSE falls short in competing with advanced water purification and electrolysis technologies. Even with major technological progress, DSE, used as a preliminary treatment, manages to cut costs and energy use by merely 0.1%, presenting considerable difficulties with minimal advantages. Therefore, in producing H_2, upcoming methods for treating seawater ought to integrate desalination processes with conventional electrolysis techniques.[93] Xie *et al.*[94] successfully utilized a water-repellent porous polytetrafluoroethylene (PTFE) membrane, which is both waterproof and breathable, as an interface for gas passage in seawater electrolysis. By selecting a dense potassium hydroxide solution as the self-dampening electrolyte (SDE), they integrated an *in situ* water purification mechanism. This system leverages a self-operating phase-change process within a seawater electrolysis apparatus, enabling efficient and innovative water electrolysis performance. This technique comprehensively tackles problems associated with adverse reactions and corrosion. Under real-world conditions, the test system functioned reliably for more than 3200 hours, maintaining a current density of 250 milliamps per square centimeter without any malfunctions. The approach effectively adopts a seawater direct electrolysis technique that is efficient, adaptable in size, and scalable, akin to freshwater electrolysis, while maintaining low operating expenses, thereby presenting vast opportunities for practical application.

Given these advancements, it is crucial to consider the broader context of seawater treatment technologies, particularly desalination. Over the past decades, the energy consumption of seawater reverse osmosis (SWRO) plants has declined considerably, from approximately 9–10 kW h m^{-3} to less than 3 kW h m^{-3}, while the cost of desalinated water has decreased from $2.2 per m^3 to below $0.6 per m^3.[95,96] Given these insights, it is essential to assess the role played by desalination within the broader framework of hydrogen production. While SWRO is often perceived as an energy-intensive process, its impact on the overall economics of hydrogen generation remains minimal. This becomes particularly evident when integrating SWRO with proton exchange membrane (PEM) water electrolysis, where the primary energy demand and cost contributions stem from the electrolysis process itself.

A cost analysis was performed for integrating seawater treatment with traditional PEM electrolysis. A typical PEM electrolysis system requires ~10 kg of water to produce 1 kg of H_2,

Table 4 Comparison of specific energy consumption by polymer electrolyte membrane water electrolysis and seawater desalination

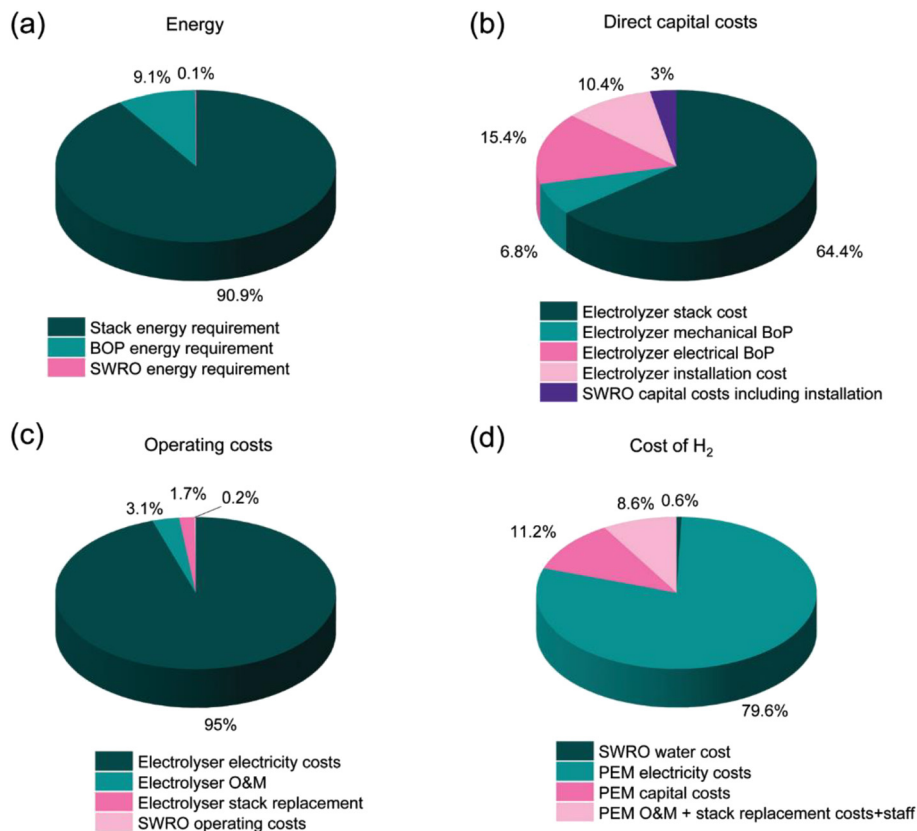| Ref.	Water demand ($kg_{H2O}\ kg_{H2}^{-1}$)	SEC of water demand *via* desalination (kW h kg_{H2}^{-1})	SEC for H_2 generation (kW h kg_{H2}^{-1})	Percentage energy for water demand (%)
Beswick *et al.*[90]	9.0	0.03	63.97	0.05
Lampert *et al.*[91]	25.7	0.09	63.97	0.14
Hydrogenics[92]	11.126	0.04	74.54	0.05
Siemens[92]	16.689	0.06	61.86	0.09

Fig. 23 Breakdown of the (a) daily energy requirement, (b) direct capital costs, (c) operating costs and (d) levelized cost of hydrogen for a SWRO–PEM electrolysis plant operating at 50 tons H_2 per day capacity.[97]

equating to ~500 m^3 per day of SWRO water for a 50 ton per day H_2 plant. The energy breakdown (Fig. 23a) shows SWRO energy demand (~0.1% of the total) is negligible compared with electrolysis. Producing 10 kg of H_2 requires ~55.44 kW h, while desalinating the same amount of water consumes only 0.03 kW h.[95] Capital expenditure for a 50 ton per day H_2 plant is ~$460 per kW installed, with 26% for balance of plant costs (Fig. 23b). SWRO capital costs depend on technology, location, and scale, but only contribute ~3% of the total capital cost. Operational costs (Fig. 23c) show PEM's electricity costs dominate (~95%), whereas SWRO accounts for ~ 0.2%, assuming electricity at $0.05 kW per h. Including SWRO raises H_2 production costs slightly, from $3.81 per kg to $3.83 per kg (Fig. 23d). These analyses indicate that SWRO's energy, capital, and operational costs are minimal compared with PEM electrolysis, making SWRO integration a low-cost solution for hydrogen production. Therefore, investing in SWRO–PEM systems offers a practical and deployable near-term pathway, compared with the uncertainties of direct seawater electrolysis.[97]

7 Summary and prospect

Efficient hydrogen production from seawater faces significant challenges, primarily requiring the development of high-performance electrocatalysts made from abundant, cost-effective materials to replace precious metals. Such alternatives can reduce energy barriers, lower overpotentials, and enhance reaction rates, thereby improving seawater electrolysis efficiency. However, most reported catalysts operate at low current densities (1–100 mA cm^{-2}), falling short of industrial demands. This review discusses HER and OER mechanisms, offering strategies for designing HER, OER, and bifunctional electrocatalysts. Key performance parameters—overpotential, Tafel slope, turnover frequency, Faraday efficiency, and stability—are also examined. Additionally, it highlights recent catalysts, including transition metal compounds and alloys, capable of sustaining industrial-grade current densities (500–1000 mA cm^{-2}) with excellent durability. Design strategies for scalable, durable, and stable catalysts are emphasized to advance industrial seawater electrolysis.

Significant progress has been made in developing electrocatalysts for industrial-grade seawater electrolysis, but challenges remain that require further systematic research. Impurities in natural seawater, including dissolved cations and microorganisms, necessitate electrocatalysts with selectively permeable layers and added electrolytes to stabilize pH at the cathode. Additionally, chloride ions interfere with the OER at the anode, requiring highly selective catalysts to suppress Cl$^-$ interference and prioritize oxygen evolution for industrial-scale hydrogen production. Excessive bubble generation compromises long-term stability, highlighting the need for binder-

free, self-supporting electrocatalysts to enhance durability. Optimizing superhydrophilic or superhydrophobic properties is crucial for efficient proton transport and bubble detachment, improving gas evolution stability.

Seawater electrolysis catalysts have been extensively studied in laboratory-scale electrolyzers, but their application in seawater electrolysis devices remains limited, necessitating research on optimizing electrolyzer configurations. Proton exchange membrane and anion exchange membrane water electrolyzers offer advantages over conventional alkaline electrolyzers, such as compact designs, low ionic resistance, and reduced energy costs. PEMWE operates under acidic conditions with mature technology for commercial-scale use, but developing durable, Ir-free OER catalysts for industrial-grade current densities remains challenging. AEMWE, suited for alkaline environments, offers a broader range of OER electrocatalysts but requires improvements in membrane stability, support structure, and bubble management for long-term operation. Integrating electrolyzers with green energy systems is also recommended to lower hydrogen production costs.

Techno-economic analysis is vital for evaluating the feasibility and scalability of hydrogen production *via* water electrolysis at industrial-grade current densities. Advances in catalyst development and device design have reduced energy consumption and costs, but challenges persist, including high electrolyzer capital costs, energy losses from overpotentials, and the need for durable, efficient materials under high-current operations. Operational costs, such as energy input, water desalination, and maintenance, further add to expenses, emphasizing cost optimization as a priority. Future TEA should focus on integrating renewable energy to lower costs and enhance sustainability. Innovations in low-cost, Earth-abundant catalysts, along with improved membranes and electrode designs, are crucial for boosting efficiency and reducing degradation. Scaling from lab prototypes to industrial systems requires balancing efficiency, durability, and economic viability. Addressing these challenges can establish electrolysis as a key technology in the global energy transition.

Data availability

The data supporting the findings of this study are included within the article and its ESI. Any additional information or datasets used or analyzed during this study are available from the corresponding author upon reasonable request.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

References

1 H. Yang, M. Driess and P. W. Menezes, *Adv. Energy Mater.*, 2021, **11**, 2102074.

2 L. Yang, L. Zhou, M. Wang, Y. Qi, D. Guo, H. Li, X. Chen and S. Wang, *Adv. Mater. Interfaces*, 2022, **9**, 2201486.

3 H. Hu, X. Wang, J. P. Attfield and M. Yang, *Chem. Soc. Rev.*, 2024, **53**, 163–203.

4 Z. Wang, W. Xu, X. Chen, Y. Peng, Y. Song, C. Lv, H. Liu, J. Sun, D. Yuan and X. Li, *Adv. Funct. Mater.*, 2019, **29**, 1902875.

5 F. Yang, Y. Luo, Q. Yu, Z. Zhang, S. Zhang, Z. Liu, W. Ren, H. M. Cheng, J. Li and B. Liu, *Adv. Funct. Mater.*, 2021, **31**, 2010367.

6 C. Yang, K. Dong, L. Zhang, X. He, J. Chen, S. Sun, M. Yue, H. Zhang, M. Zhang and D. Zheng, *Inorg. Chem.*, 2023, **62**, 7976–7981.

7 X. Kang, F. Yang, Z. Zhang, H. Liu, S. Ge, S. Hu, S. Li, Y. Luo, Q. Yu and Z. Liu, *Nat. Commun.*, 2023, **14**, 3607.

8 H. Liu, R. Xie, Y. Luo, Z. Cui, Q. Yu, Z. Gao, Z. Zhang, F. Yang, X. Kang and S. Ge, *Nat. Commun.*, 2022, **13**, 6382.

9 Q. Wang, Y. Cheng, H. B. Tao, Y. Liu, X. Ma, D. S. Li, H. B. Yang and B. Liu, *Angew. Chem.*, 2023, **135**, e202216645.

10 S. Li, E. Li, X. An, X. Hao, Z. Jiang and G. Guan, *Nanoscale*, 2021, **13**, 12788–12817.

11 S. Wang, A. Lu and C. J. Zhong, *Nano Convergence*, 2021, **8**, 4.

12 Z. P. Wu, X. F. Lu, S. Q. Zang and X. W. Lou, *Adv. Funct. Mater.*, 2020, **30**, 1910274.

13 X. Lyu and A. Serov, *Ind. Chem. Mater.*, 2023, **1**, 475–485.

14 H. Jin, X. Wang, C. Tang, A. Vasileff, L. Li, A. Slattery and S. Z. Qiao, *Adv. Mater.*, 2021, **33**, 2007508.

15 M. Auinger, I. Katsounaros, J. C. Meier, S. O. Klemm, P. U. Biedermann, A. A. Topalov, M. Rohwerder and K. J. J. Mayrhofer, *Phys. Chem. Chem. Phys.*, 2011, **13**, 16384–16394.

16 L. Wu, F. Zhang, S. Song, M. Ning, Q. Zhu, J. Zhou, G. Gao, Z. Chen, Q. Zhou and X. Xing, *Adv. Mater.*, 2022, **34**, 2201774.

17 X. Xiao, L. Yang, W. Sun, Y. Chen, H. Yu, K. Li, B. Jia, L. Zhang and T. Ma, *Small*, 2022, **18**, 2105830.

18 J. G. Vos, Z. Liu, F. D. Speck, N. Perini, W. Fu, S. Cherevko and M. T. M. Koper, *ACS Catal.*, 2019, **9**, 8561–8574.

19 W. Tong, M. Forster, F. Dionigi, S. Dresp, R. Sadeghi Erami, P. Strasser, A. J. Cowan and P. Farràs, *Nat. Energy*, 2020, **5**, 367–377.

20 F. Dionigi, T. Reier, Z. Pawolek, M. Gliech and P. Strasser, *ChemSusChem*, 2016, **9**, 962–972.

21 Y. Luo, Z. Zhang, F. Yang, J. Li, Z. Liu, W. Ren, S. Zhang and B. Liu, *Energy Environ. Sci.*, 2021, **14**, 4610–4619.

22 X.-Y. Zhang, Y.-R. Zhu, Y. Chen, S.-Y. Dou, X.-Y. Chen, B. Dong, B.-Y. Guo, D.-P. Liu, C.-G. Liu and Y.-M. Chai, *Chem. Eng. J.*, 2020, **399**, 125831.

23 L. Zhang, Z. Shi, Y. Lin, F. Chong and Y. Qi, *Front. Chem.*, 2022, **10**, 866415.

24 D. Liu, Y. Cai, X. Wang, Y. Zhuo, X. Sui, H. Pan and Z. Wang, *Energy Environ. Sci.*, 2024, **17**, 6897–6942.

25 M. F. Lagadec and A. Grimaud, *Nat. Mater.*, 2020, **19**, 1140–1150.

26 K. Jiang, W. Liu, W. Lai, M. Wang, Q. Li, Z. Wang, J. Yuan, Y. Deng, J. Bao and H. Ji, *Inorg. Chem.*, 2021, **60**, 17371–17378.

27 L. Xiu, W. Pei, S. Zhou, Z. Wang, P. Yang, J. Zhao and J. Qiu, *Adv. Funct. Mater.*, 2020, **30**, 1910028.

28 H. Jin, J. Xu, H. Liu, H. Shen, H. Yu, M. Jaroniec, Y. Zheng and S.-Z. Qiao, *Sci. Adv.*, 2023, **9**, eadi7755.

29 X. Yu, Z.-Y. Yu, X.-L. Zhang, Y.-R. Zheng, Y. Duan, Q. Gao, R. Wu, B. Sun, M.-R. Gao and G. Wang, *J. Am. Chem. Soc.*, 2019, **141**, 7537–7543.

30 L. A. King, M. A. Hubert, C. Capuano, J. Manco, N. Danilovic, E. Valle, T. R. Hellstern, K. Ayers and T. F. Jaramillo, *Nat. Nanotechnol.*, 2019, **14**, 1071–1074.

31 Y. Luo, Z. Zhang, M. Chhowalla and B. Liu, *Adv. Mater.*, 2022, **34**, 2108133.

32 W. Xu, G. Fan, S. Zhu, Y. Liang, Z. Cui, Z. Li, H. Jiang, S. Wu and F. Cheng, *Adv. Funct. Mater.*, 2021, **31**, 2107333.

33 Z. Yan, H. Sun, X. Chen, H. Liu, Y. Zhao, H. Li, W. Xie, F. Cheng and J. Chen, *Nat. Commun.*, 2018, **9**, 2373.

34 P. Liu, B. Chen, C. Liang, W. Yao, Y. Cui, S. Hu, P. Zou, H. Zhang, H. J. Fan and C. Yang, *Adv. Mater.*, 2021, **33**, 2007377.

35 R. Rossi, D. M. Hall, L. Shi, N. R. Cross, C. A. Gorski, M. A. Hickner and B. E. Logan, *Energy Environ. Sci.*, 2021, **14**, 6041–6049.

36 L. Li, G. Zhang, B. Wang, D. Zhu, D. Liu, Y. Liu and S. Yang, *ACS Appl. Mater. Interfaces*, 2021, **13**, 37152–37161.

37 C. Liang, P. Zou, A. Nairan, Y. Zhang, J. Liu, K. Liu, S. Hu, F. Kang, H. J. Fan and C. Yang, *Energy Environ. Sci.*, 2020, **13**, 86–95.

38 L. Yu, L. Wu, B. McElhenny, S. Song, D. Luo, F. Zhang, Y. Yu, S. Chen and Z. Ren, *Energy Environ. Sci.*, 2020, **13**, 3439–3446.

39 T. Haq and Y. Haik, *ACS Sustainable Chem. Eng.*, 2022, **10**, 6622–6632.

40 H. Ge, G. Li, J. Shen, W. Ma, X. Meng and L. Xu, *Appl. Catal., B*, 2020, **275**, 119104.

41 L. Wang, Y. Li, X. Yin, Y. Wang, A. Song, Z. Ma, X. Qin and G. Shao, *ACS Sustainable Chem. Eng.*, 2017, **5**, 7993–8003.

42 W.-J. Ong, L.-L. Tan, Y. H. Ng, S.-T. Yong and S.-P. Chai, *Chem. Rev.*, 2016, **116**, 7159–7329.

43 F. Zhang, Y. Liu, L. Wu, M. Ning, S. Song, X. Xiao, V. G. Hadjiev, D. E. Fan, D. Wang and L. Yu, *Mater. Today Phys.*, 2022, **27**, 100841.

44 Y. S. Park, J. Lee, M. J. Jang, J. Yang, J. Jeong, J. Park, Y. Kim, M. H. Seo, Z. Chen and S. M. Choi, *J. Mater. Chem. A*, 2021, **9**, 9586–9592.

45 H. Komiya, T. Shinagawa and K. Takanabe, *ChemSusChem*, 2022, **15**, e202201088.

46 C. Wang, M. Zhu, Z. Cao, P. Zhu, Y. Cao, X. Xu, C. Xu and Z. Yin, *Appl. Catal., B*, 2021, **291**, 120071.

47 C. Wang, X. Shao, J. Pan, J. Hu and X. Xu, *Appl. Catal., B*, 2020, **268**, 118435.

48 L. Yu, L. Wu, B. McElhenny, S. Song, D. Luo, F. Zhang, Y. Yu, S. Chen and Z. Ren, *Energy Environ. Sci.*, 2020, **13**, 3439–3446.

49 M. Yu, J. Li, F. Liu, J. Liu, W. Xu, H. Hu, X. Chen, W. Wang and F. Cheng, *J. Energy Chem.*, 2022, **72**, 361–369.

50 Y. Kuang, M. J. Kenney, Y. Meng, W.-H. Hung, Y. Liu, J. E. Huang, R. Prasanna, P. Li, Y. Li, L. Wang, M.-C. Lin, M. D. McGehee, X. Sun and H. Dai, *Proc. Natl. Acad. Sci. U. S. A.*, 2019, **116**, 6624–6629.

51 L. Qi, A. Li, M. Wang, Y. Zhang, K. Zhang and X. Li, *Adv. Mater. Interfaces*, 2022, **9**, 2101720.

52 J. Bian, Z. Song, X. Li, Y. Zhang and C. Cheng, *Nanoscale*, 2020, **12**, 8443–8452.

53 C. Chen, Y. Tuo, Q. Lu, H. Lu, S. Zhang, Y. Zhou, J. Zhang, Z. Liu, Z. Kang, X. Feng and D. Chen, *Appl. Catal., B*, 2021, **287**, 119953.

54 Z. Lei, X. Jin, X. Li, Y. Liu, J. Liu, S. Jiao and R. Cao, *J. Energy Chem.*, 2022, **65**, 505–513.

55 L. Li, G. Zhang, B. Wang, D. Zhu, D. Liu, Y. Liu and S. Yang, *ACS Appl. Mater. Interfaces*, 2021, **13**, 37152–37161.

56 T. U. Haq and Y. Haik, *ACS Sustainable Chem. Eng.*, 2022, **10**, 6622–6632.

57 F. Zhang, Y. Liu, L. Wu, M. Ning, S. Song, X. Xiao, V. G. Hadjiev, D. E. Fan, D. Wang, L. Yu, S. Chen and Z. Ren, *Mater. Today Phys.*, 2022, **27**, 100841.

58 Y. S. Park, J. Lee, M. J. Jang, J. Yang, J. Jeong, J. Park, Y. Kim, M. H. Seo, Z. Chen and S. M. Choi, *J. Mater. Chem. A*, 2021, **9**, 9586–9592.

59 H. Komiya, T. Shinagawa and K. Takanabe, *ChemSusChem*, 2022, **15**, e202201088.

60 C. Wang, M. Zhu, Z. Cao, P. Zhu, Y. Cao, X. Xu, C. Xu and Z. Yin, *Appl. Catal., B*, 2021, **291**, 120071.

61 L. Guo, J. Chi, J. Zhu, T. Cui, J. Lai and L. Wang, *Appl. Catal., B*, 2023, **320**, 121977.

62 N. Q. Tran, B. T. N. Le, T. N.-M. Le, L. T. Duy, T. B. Phan, Y. Hong, T.-K. Truong, T. L. H. Doan, J. Yu and H. Lee, *J. Phys. Chem. Lett.*, 2022, **13**, 8192–8199.

63 F. Yang, Y. Luo, Q. Yu, Z. Zhang, S. Zhang, Z. Liu, W. Ren, H.-M. Cheng, J. Li and B. Liu, *Adv. Funct. Mater.*, 2021, **31**, 2010367.

64 C. Chai, J. Yang, C. Jiang, L. Liu and J. Xi, *ACS Appl. Energy Mater.*, 2022, **5**, 2909–2917.

65 C. Wei, Y. Sun, G. G. Scherer, A. C. Fisher, M. Sherburne, J. W. Ager and Z. J. Xu, *J. Am. Chem. Soc.*, 2020, **142**, 7765–7775.

66 G. Li, S. Feng, J. Li, P. Deng, X. Tian, C. Wang and Y. Hua, *Chin. J. Struct. Chem.*, 2022, **41**, 2207068–2207073.

67 Z. Yu, J. Xu, L. Meng and L. Liu, *J. Mater. Chem. A*, 2021, **9**, 22248–22253.

68 F. Dionigi, T. Reier, Z. Pawolek, M. Gliech and P. Strasser, *ChemSusChem*, 2016, **9**, 962–972.

69 C. Li, W. Hong, Q. Cai and C. Jian, *ACS Appl. Mater. Interfaces*, 2022, **14**, 30683–30691.

70 H. Lv, C. Fu, J. Fan, Y. Zhang and W. Hao, *J. Colloid Interface Sci.*, 2022, **626**, 384–394.

71 Z. Wang, W. Xu, K. Yu, Y. Feng and Z. Zhu, *Nanoscale*, 2020, **12**, 6176–6187.

72 X. Du, M. Tan, T. Wei, H. Kobayashi, J. Song, Z. Peng, H. Zhu, Z. Jin, R. Li and W. Liu, *Chem. Eng. J.*, 2023, **452**, 139404.

73 J. Peng, Y. Li, Z. Chen, G. Liang, S. Hu, T. Zhou, F. Zheng, Q. Pan, H. Wang, Q. Li, J. Liu and Z. Guo, *ACS Nano*, 2021, **15**, 11607–11618.

74 C. Fu, S. Weng, J. Fan, Y. Zhang, Y. Guo and W. Hao, *Chem. Eng. J.*, 2022, **430**, 132881.

75 W. Xu, G. Fan, S. Zhu, Y. Liang, Z. Cui, Z. Li, H. Jiang, S. Wu and F. Cheng, *Adv. Funct. Mater.*, 2021, **31**, 2107333.

76 D. Wu, D. Chen, J. Zhu and S. Mu, *Small*, 2021, **17**, 2102777.

77 D. Kong, C. Meng, Y. Wang, X. Chen, J. Zhang, L. Zhao, J. Ji, L. Zhang and Y. Zhou, *Appl. Catal., B*, 2024, **343**, 123578.

78 J. Guo, Y. Zheng, Z. Hu, C. Zheng, J. Mao, K. Du, M. Jaroniec, S.-Z. Qiao and T. Ling, *Nat. Energy*, 2023, **8**, 264–272.

79 J. Chang, G. Wang, Z. Yang, B. Li, Q. Wang, R. Kuliiev, N. Orlovskaya, M. Gu, Y. Du, G. Wang and Y. Yang, *Adv. Mater.*, 2021, **33**, 2101425.

80 Z. Chen, Q. Li, H. Xiang, Y. Wang, P. Yang, C. Dai, H. Zhang, W. Xiao, Z. Wu and L. Wang, *Inorg. Chem. Front.*, 2023, **10**, 1493–1500.

81 D. Yao, W. Hao, S. Weng, M. Hou, W. Cen, G. Li, Z. Chen and Y. Li, *Small*, 2022, **18**, 2106868.

82 Z. Yu, Y. Li, V. Martin-Diaconescu, L. Simonelli, J. Ruiz Esquius, I. Amorim, A. Araujo, L. Meng, J. L. Faria and L. Liu, *Adv. Funct. Mater.*, 2022, **32**, 2206138.

83 M. Ning, F. Zhang, L. Wu, X. Xing, D. Wang, S. Song, Q. Zhou, L. Yu, J. Bao, S. Chen and Z. Ren, *Energy Environ. Sci.*, 2022, **15**, 3945–3957.

84 B. Wang, M. Lu, D. Chen, Q. Zhang, W. Wang, Y. Kang, Z. Fang, G. Pang and S. Feng, *J. Mater. Chem. A*, 2021, **9**, 13562–13569.

85 T. Van Tam, K. C. Bhamu, M. J. Kim, S. G. Kang, J. S. Chung, S. H. Hur and W. M. Choi, *Chem. Eng. J.*, 2024, **480**, 148190.

86 H. Fei, R. Liu, T. Liu, M. Ju, J. Lei, Z. Wang, S. Wang, Y. Zhang, W. Chen, Z. Wu, M. Ni and J. Wang, *Adv. Mater.*, 2024, **36**, 2309211.

87 J. E. Park, S. Y. Kang, S.-H. Oh, J. K. Kim, M. S. Lim, C.-Y. Ahn, Y.-H. Cho and Y.-E. Sung, *Electrochim. Acta*, 2019, **295**, 99–106.

88 L. Wan, Z. Xu, Q. Xu, M. Pang, D. Lin, J. Liu and B. Wang, *Energy Environ. Sci.*, 2023, **16**, 1384–1430.

89 G. Zang, E. J. Graham and D. Mallapragada, *Int. J. Hydrogen Energy*, 2024, **49**, 1288–1303.

90 R. R. Beswick, A. M. Oliveira and Y. Yan, *ACS Energy Lett.*, 2021, **6**, 3167–3169.

91 D. J. Lampert, H. Cai and A. Elgowainy, *Energy Environ. Sci.*, 2016, **9**, 787–802.

92 A. T. Mayyas, M. F. Ruth, B. S. Pivovar, G. Bender and K. B. Wipke, *Manufacturing Cost Analysis for Proton Exchange Membrane Water Electrolyzers*, National Renewable Energy Laboratory (NREL), Golden, CO (United States), United States, 2019.

93 B. Lee, L. Wang, Z. Wang, N. J. Cooper and M. Elimelech, *Energy Environ. Sci.*, 2023, **16**, 714–722.

94 H. Xie, Z. Zhao, T. Liu, Y. Wu, C. Lan, W. Jiang, L. Zhu, Y. Wang, D. Yang and Z. Shao, *Nature*, 2022, **612**, 673–678.

95 N. Ghaffour, S. Lattemann, T. Missimer, K. C. Ng, S. Sinha and G. Amy, *Appl. Energy*, 2014, **136**, 1155–1165.

96 N. Ghaffour, T. M. Missimer and G. L. Amy, *Desalination*, 2013, **309**, 197–207.

97 M. Khan, T. Al-Attas, S. Roy, M. M. Rahman, N. Ghaffour, V. Thangadurai, S. Larter, J. Hu, P. M. Ajayan and M. G. Kibria, *Energy Environ. Sci.*, 2021, **14**, 4831–4839.