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# Highly sensitive and room temperature detection of ultra-low concentrations of O<sub>3</sub> using self-powered sensing elements of Cu<sub>2</sub>O nanocubes

E. Petromichelaki,<sup>a</sup> E. Gagaoudakis,<sup>ab</sup> K. Moschovis,<sup>a</sup> L. Tsetseris,<sup>id c</sup>  
T. D. Anthopoulos,<sup>id d</sup> G. Kiriakidis<sup>ab</sup> and V. Binas<sup>id \*ab</sup>

The fundamental development of the design of novel self-powered ozone sensing elements, operating at room temperature, based on p-type metal oxides paves the way to a new class of low cost, highly promising gas sensing devices. In this work, p-type Cu<sub>2</sub>O nanocubes were synthesized by a simple solution-based method and tested as a self-powered ozone sensing element, at room temperature (25 °C) for the first time. Highly crystalline Cu<sub>2</sub>O nanocubes with 30 nm size were characterized by X-ray Diffraction (XRD), Scanning Electron Microscopy (SEM) and Transmission Electron Microscopy (TEM). Self-powered sensing elements of Cu<sub>2</sub>O nanocubes have been successfully fabricated by deposition of Cu<sub>2</sub>O nanocubes on interdigitated electrodes (IDEs) consisting of two connection tracks with 500 digits and a gap of 5 μm in order to investigate their response to ozone at room temperature. The experimental results showed that the use of nanocubes as sensing elements was suitable for detecting ultra-low concentrations of O<sub>3</sub> down to 10 ppb at room temperature with very high sensitivity (28%) and a very low response/recovery time. The reversible sensing process of the relatively weak binding of O<sub>3</sub> species by trapping sites on Cu<sub>2</sub>O facets with increased oxygen content was studied by using density functional theory (DFT) calculations.

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## Introduction

Some of the most common pollutant gases in air are NO<sub>2</sub>, O<sub>3</sub>, CO, SO<sub>2</sub> and NO. Among them, ozone (O<sub>3</sub>) is closely related to human life since it is widely used in a numerous applications such as in food storage and in the pharmaceutical, textile and chemical industries. Moreover, it works as a protective layer in the Earth's stratosphere for human health, absorbing the harmful ultraviolet (UV) radiation. On the other hand, according to the World Health Organization (WHO), excess exposure to ozone concentrations of more than 1 parts per million (ppm) can cause breathing problems, trigger asthma, reduce lung function and induce lung diseases and induce the burning eyes symptom. Thus, it is apparent that detection of ozone at ultra-low concentrations (*i.e.* low ppb levels) is of great importance.<sup>1</sup>

The use of metal oxide semiconductors as sensing elements for monitoring air quality has emerged as a low cost alternative

due to fast response/recovery rates and low detection limits.<sup>2–4</sup> In particular, cuprous oxide (Cu<sub>2</sub>O) as a p-type semiconductor, with a narrow direct energy band gap of  $E_g = 2.17$  eV, has been tested as gas sensor material in recent years and highlighted due to its good stability and high conductivity. In addition, Cu<sub>2</sub>O is non-toxic and copper is earth abundant and hence inexpensive. To date, numerous micro- and nano-structures of octahedral<sup>4,5</sup> and polyhedral<sup>6</sup> Cu<sub>2</sub>O have been reported, with a variety of morphologies, such as nanocubes,<sup>7,8</sup> spheres,<sup>9</sup> nanowires,<sup>10</sup> and quasi-spherical shapes,<sup>11</sup> using solution-based synthetic methods. Cu<sub>2</sub>O films have been also produced *via* a variety of methods including thermal oxidation, magnetron sputtering, spray pyrolysis, *etc.*<sup>12–15</sup> In the field of gas-sensing, Cu<sub>2</sub>O has great potential in detecting various pollutants in their gaseous form, due to its significant surface reactivity with both reducing (ethanol, gasoline, CO, and H<sub>2</sub>S)<sup>3,5,7,8,16</sup> and oxidizing gases (NO<sub>2</sub>, CO<sub>2</sub>, CH<sub>4</sub>, and O<sub>3</sub>).<sup>4,6,7,9,10,17,18</sup> However, the working temperature of the sensors fabricated using Cu<sub>2</sub>O is significantly higher than room temperature (25 °C).<sup>4,6,9–13</sup>

Considering the detection of O<sub>3</sub>, the majority of studies are related to composites with other metals such as CuAlO<sub>2</sub> and CuCrO<sub>2</sub> tested against a variety of gas concentrations and at different working temperatures. To the best of our knowledge, Ouali *et al.* were the only report on the use of Cu<sub>2</sub>O for 10 to 1000 ppb level sensing of O<sub>3</sub> at an operating temperature of 175 °C.<sup>17</sup> Moreover, Zhou *et al.* synthesized both nanocrystals

<sup>a</sup>Institute of Electronic Structure and Laser (IESL), FORTH, P. O. Box 1527, Vasilika Vouton, GR-71110 Heraklion, Greece. E-mail: binasbill@iesl.forth.gr

<sup>b</sup>Crete Center for Quantum Complexity & Nanotechnology (CCQCN), Department of Physics, University of Crete, GR-71003, Heraklion, Greece

<sup>c</sup>Department of Physics, National Technical University of Athens, GR-15780 Athens, Greece

<sup>d</sup>Physical Science and Engineering Division, King Abdullah University of Science and Technology, Saudi Arabia







Fig. 1 Conductivity-based system with a gas test chamber.

### Density functional theory (DFT) calculations

First-principles spin-polarized calculations were performed with the Density Functional Theory (DFT) code Quantum Espresso,<sup>24</sup> projector-augmented waves<sup>25</sup> to model the interactions between valence electrons and ionic cores, the PW91 exchange–correlation (xc) functional<sup>26</sup> and the DFT-D2 method<sup>27</sup> to take into account van der Waals interactions. The energy cutoff of the plane wave basis was set at 75 Rydberg (Ry) and integrations in reciprocal space used  $2 \times 2 \times 1$  or  $4 \times 4 \times 1$  Monkhorst–Pack  $k$ -point grids<sup>28</sup> or tetrahedron sampling,<sup>29</sup> respectively, for the determination of total energies or electronic densities of states. The (001) surfaces of  $\text{Cu}_2\text{O}$  were modeled as slabs with 7 layers and  $2 \times 2$  supercells in the lateral directions (with at least 86 atoms). The relaxation of structures was terminated when the energy converged within 10–6 Ry. We also obtained the energies for the highest occupied (HOMO) and lowest unoccupied (LUMO) molecular orbitals of an  $\text{O}_3$  molecule with the hybrid B3LYP xc-functional<sup>32</sup> and the 6-311+G\* Gaussian basis set (as implemented<sup>30</sup> in the NWChem code).

## Results and discussion

$\text{Cu}_2\text{O}$  nanocubes were characterized using a powder X-ray diffractometer (XRD). Fig. 2a shows the XRD pattern of  $\text{Cu}_2\text{O}$  nanocubes and the standard  $\text{Cu}_2\text{O}$  card. All the diffraction peaks are consistent with the standard card (JCPDS card no. 05-0667) of cubic phase  $\text{Cu}_2\text{O}$  attributed to the (110), (111), (200), (220), (311) and (222) crystal planes. Specifically, the strong and

sharp peaks indicate that the obtained  $\text{Cu}_2\text{O}$  nanocubes are highly crystalline with an average crystal size of 19.7 nm, as calculated using Scherrer's formula. After the controllable synthesis of  $\text{Cu}_2\text{O}$  nanocubes, the representative morphology and size were determined by field emission scanning electron microscopy (FESEM) and high resolution transition electron microscopy (TEM) as shown in Fig. 2b and c. From this, it can be seen that  $\text{Cu}_2\text{O}$  material is mainly composed of uniform cubes that are 30 to 35 nm in size. Highly crystalline  $\text{Cu}_2\text{O}$  nanocubes with 30 nm size were deposited on interdigitated electrodes with a good coverage as shown in Fig. 3a and b.

The two-terminal interdigitated sensor devices were tested at room temperature whilst been exposed to different concentrations of ozone in the range 10–820 ppb. The repeatability of the sensing response was also examined by repeating the measurement several times using different  $\text{Cu}_2\text{O}$  nanocube-based devices. Fig. 4 shows the evolution of the current (measured at a constant bias of 10 volts) as a function of time and exposure to different ozone concentrations. In particular, the measurement included two steps: (i) the sensor was exposed to an atmosphere containing  $\text{O}_3$  with a predetermined concentration for approximately 2 min while the current was monitored in real time; (ii) the ozone flow was turned-off and the device was exposed to ambient air (relative humidity  $\approx$  55%) for approximately 1 min. The same measuring cycle was repeated for  $\text{O}_3$  concentrations of 170, 490 and 820 ppb (Fig. 4). It is clearly evident that the current across the  $\text{Cu}_2\text{O}$  nanocube channel increases in the presence of ozone, an observation







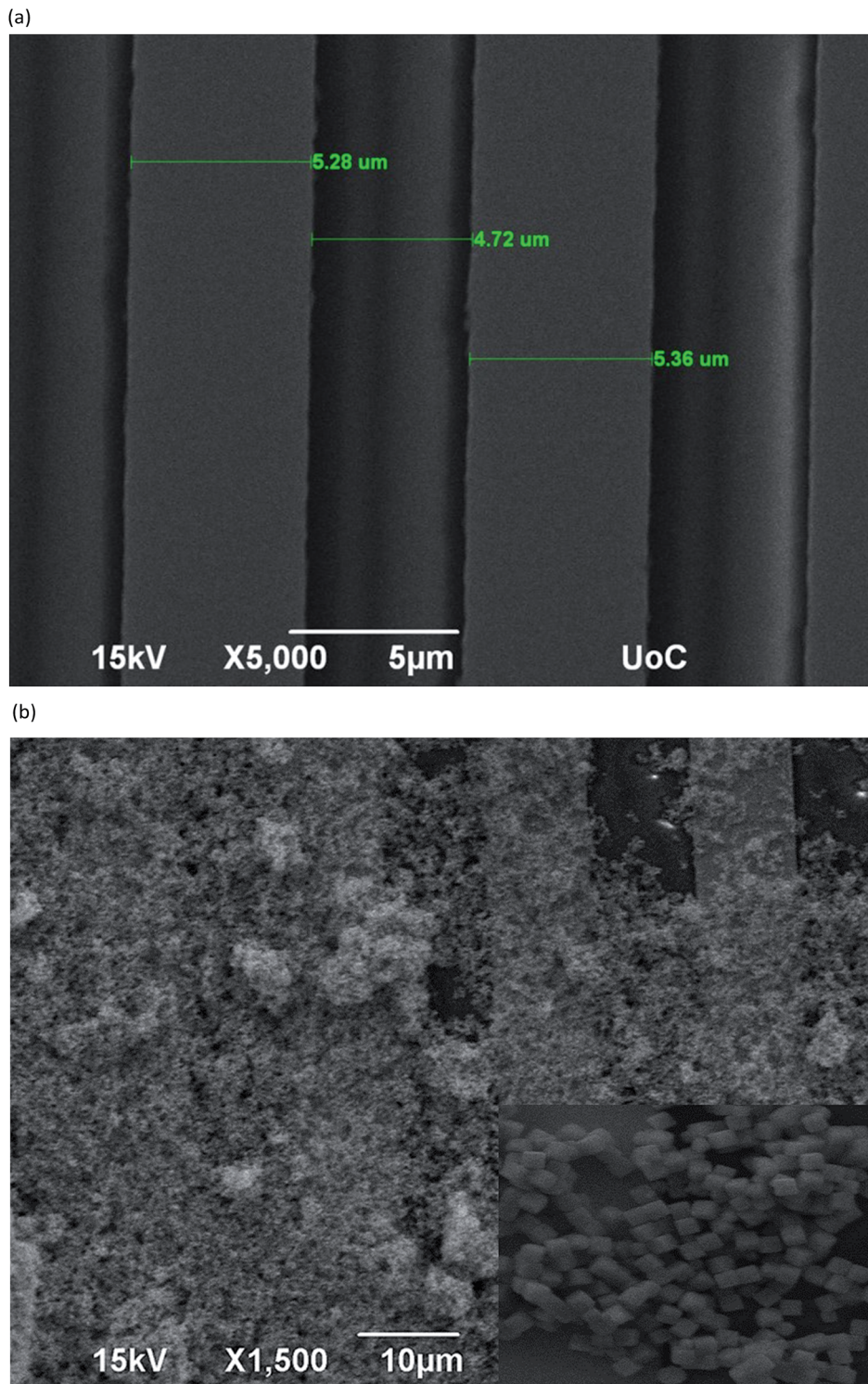


Fig. 3 (a) SEM images of the IDEs with a gap of 5  $\mu\text{m}$ . (b) SEM of coated IDEs and zoomed view of the coated IDEs with nanocubes in the inset.

the adsorption of  $\text{O}_2$  and  $\text{O}_3$  molecules on (001) surfaces, given that (001) facets dominate  $\text{Cu}_2\text{O}$  nanocubes,<sup>8</sup> while (110) and (111) terminations appear, respectively, only as truncated edges and corners.<sup>33</sup>

First, we should note that the calculated LUMO value ( $-5.65$  eV) is lower than the measured<sup>31</sup> valence band maximum ( $-5.3$

eV) of  $\text{Cu}_2\text{O}$ . Hence, any physisorbed  $\text{O}_3$  species should act as electron acceptors for  $\text{Cu}_2\text{O}$ . By the same token, ozone molecules can interact directly with the  $\text{Cu}_2\text{O}$  facets. We found that when these facets have low oxygen content (close to the value associated with the stoichiometry of  $\text{Cu}_2\text{O}$ ),  $\text{O}_2$  and  $\text{O}_3$  molecules chemisorb to configurations with large binding energies





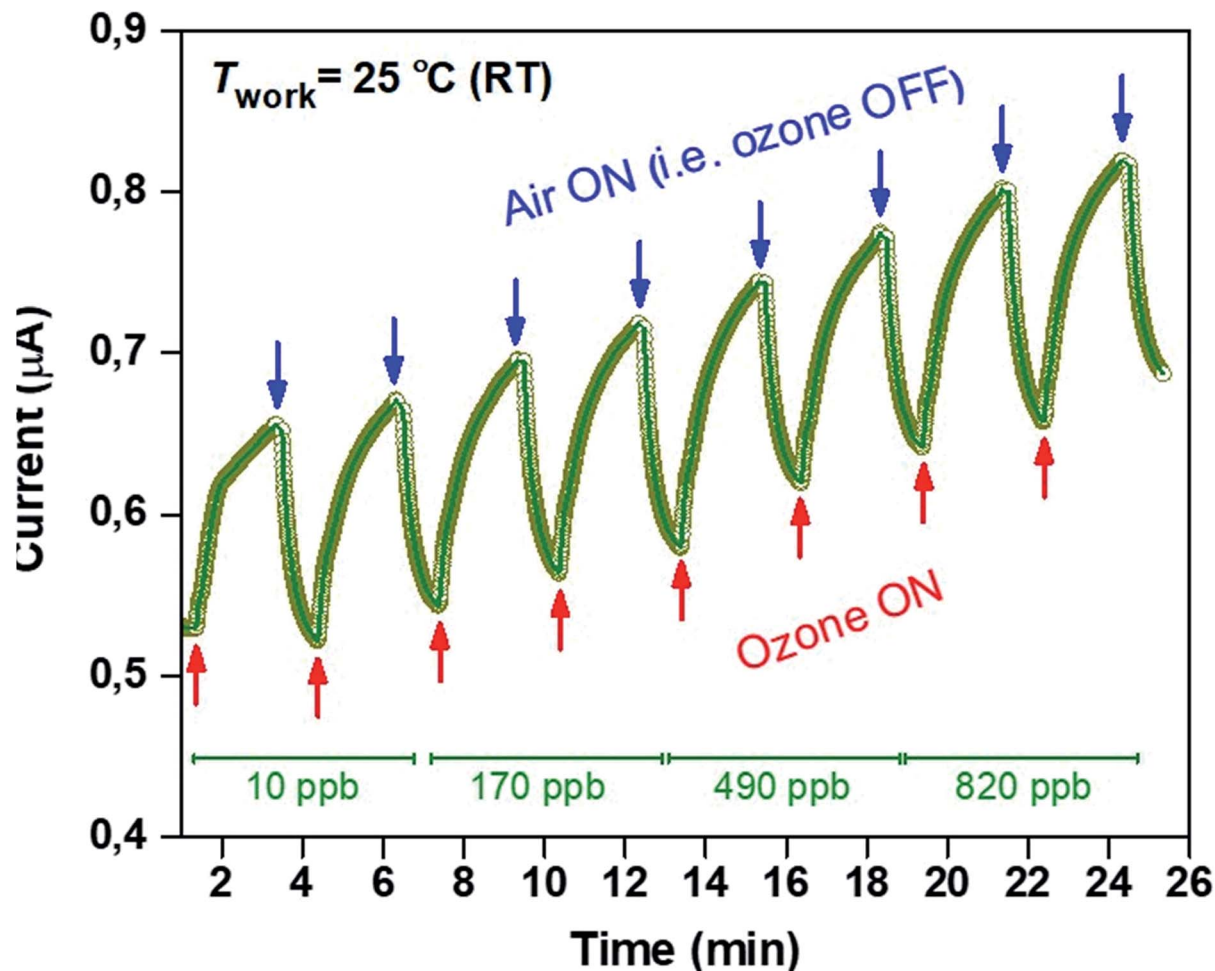


Fig. 4 Current changes at room temperature of Cu<sub>2</sub>O nanocubes due to successive increases in O<sub>3</sub> concentrations.

(in excess of 1.5 eV) and some of the deposited O species move to subsurface sites. Concurrently the Fermi level of the system moves inside its valence band consistent with p-type doping. These results are in agreement with the findings reported by Quali *et al.*<sup>17</sup>

Fig. 6 shows typical steps for the oxidation reactions taking place on Cu<sub>2</sub>O surfaces with a relatively low O content. First, O<sub>2</sub> molecules chemisorb in barrier-less processes and create species like the one shown in the configuration labelled (a) in Fig. 6. Starting from such a chemisorbed configuration, the adsorbed molecule easily breaks into two O adatoms, since the corresponding barrier is only about 0.4 eV. This low activation energy shows that the process takes place even at room temperature. Following the breakup of the molecule, the two O adatoms relax to the intermediate configuration (c) of Fig. 6, and eventually to an even more stable (by about 2 eV) freshly oxidized structure (not shown here). The oxidation reactions with an O<sub>3</sub> molecule follow similar steps, namely there is a barrier-less initial chemisorption, followed by an essentially barrier-less breakup of one of the three O atoms of the adsorbed O<sub>3</sub> species, and then by reactions akin to those depicted in Fig. 6.

On the other hand, O<sub>3</sub> chemisorption continues to take place on areas with increased oxygen content, but in selected such

cases, the result is the formation of O<sub>3</sub>-related defects with relatively low binding energies. Fig. 7 shows two such configurations, an O<sub>3</sub>-protrusion in Fig. 7a with a single Cu–O bond and an inter-Cu O<sub>3</sub>-bridge in Fig. 7b. The low binding energies of these O<sub>3</sub> configurations (0.90 eV and 0.40 eV, respectively) suggest that when the ozone supply is interrupted, some of the adsorbed O<sub>3</sub> species can be replaced by physisorbed O<sub>2</sub> molecules which are present in much larger numbers and have corresponding binding energies of 0.23 eV and 0.10 eV. With respect to electronic properties, the results of projected DOS (not shown here) show that the defects shown in Fig. 7 introduce peaks at 0.05 eV and 0.3 eV above the Fermi level of the structure, and hence they can indeed play the role of electron acceptors (though we should note that the thickness of the slab and the use of the PW91 xc-functional do not allow us to draw final conclusions on the exact position of these peaks).

The main picture that emerges from these calculations is the following. If Cu<sub>2</sub>O nanocrystals are not O-rich in the beginning, then the deposition of electronegative oxygen atoms (through, *e.g.*, the dissociation of O<sub>3</sub> molecules) contributes to the p-doping of the material. Taking into account also the experimental findings on enhanced hole conductivity, the main reactions during this first stage can be described schematically as:



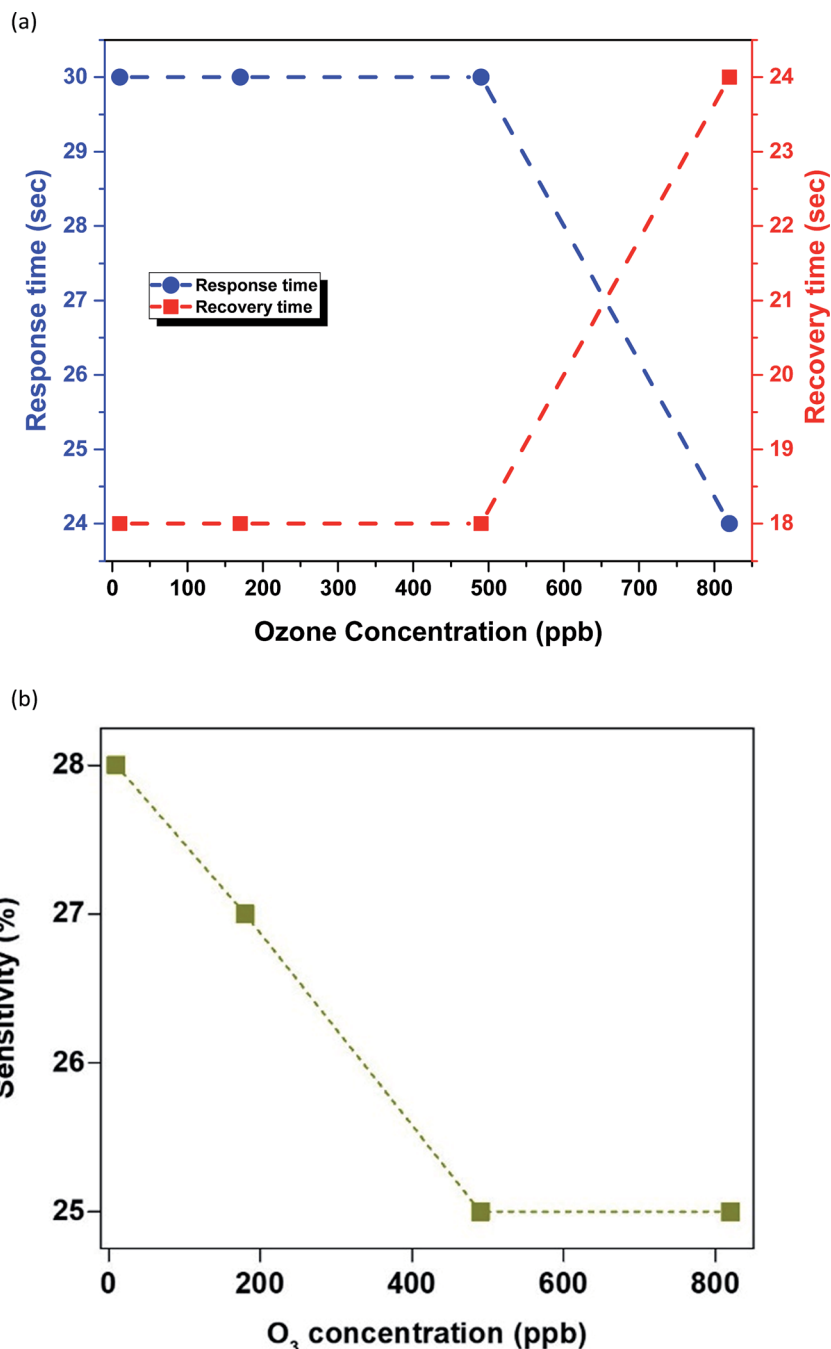
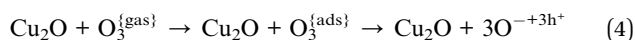
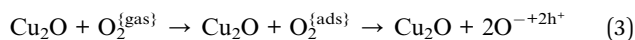
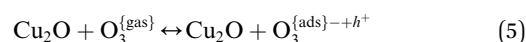


Fig. 5 (a) Response and recovery time (s) of the sensor at different ozone concentrations. (b) Sensing sensitivity as a function of ozone concentration.



When the nanocrystal facets are sufficiently oxidized (a condition which is plausible, but which also depends on the details of growth and operation), then they can selectively trap O<sub>3</sub> molecules with binding energies that are neither very large (so that the process may still be reversible) nor very small (so that it can lead to sensing of even small amounts of ozone). This second stage can be described as:



This atomic-scale picture is consistent with the broad features observed in the experiments. Nevertheless, we should stress that the high reactivity of the material (which also results in large local distortions upon oxygen uptake) leaves open the possibility that other adsorption mechanisms may also be important here. Future studies will explore, for example, the role of humidity or of surface reconstructions.



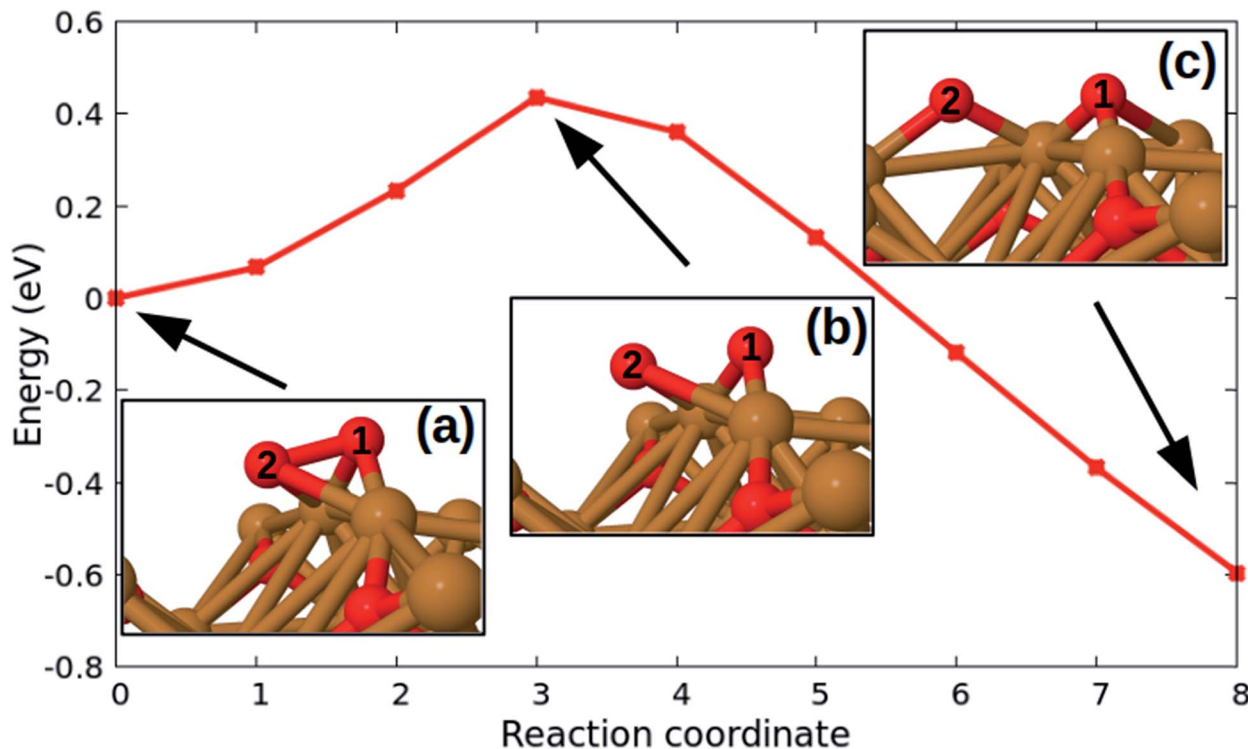


Fig. 6 Energy variation during a typical oxidation reaction on a (001)  $\text{Cu}_2\text{O}$  surface. An  $\text{O}_2$  molecule (its atoms are numbered 1 and 2) chemisorbs with a zero barrier to configuration (a), followed by its breakup to atomic O adatoms and structures (b) and (c). The transition state of the reaction is (b) and the barrier is low and equal to 0.43 eV.

On the basis of the results and analysis presented so far one can argue that there are a few plausible reasons responsible for the fact that the current does not fully recover upon re-exposure of the sensor to ambient air (Fig. 4). The first reason is that the amount of air inserted into chamber for the given recovery (reset) period ( $\sim 1$  min) is not enough to completely decrease the number of holes that are being created by the adsorption of  $\text{O}_3$  on  $\text{Cu}_2\text{O}$  during the

exposure time. Increasing the recovery period would therefore help to reduce the current closer to its original value. Another reason for the incomplete current recovery is dissociation kinetics associated with  $\text{O}_3$  desorption from the surface of  $\text{Cu}_2\text{O}$  at room temperature. This issue could potentially be addressed by operating the sensor at elevated temperature and/or through photo-assisted dissociation albeit at the cost of operational simplicity.

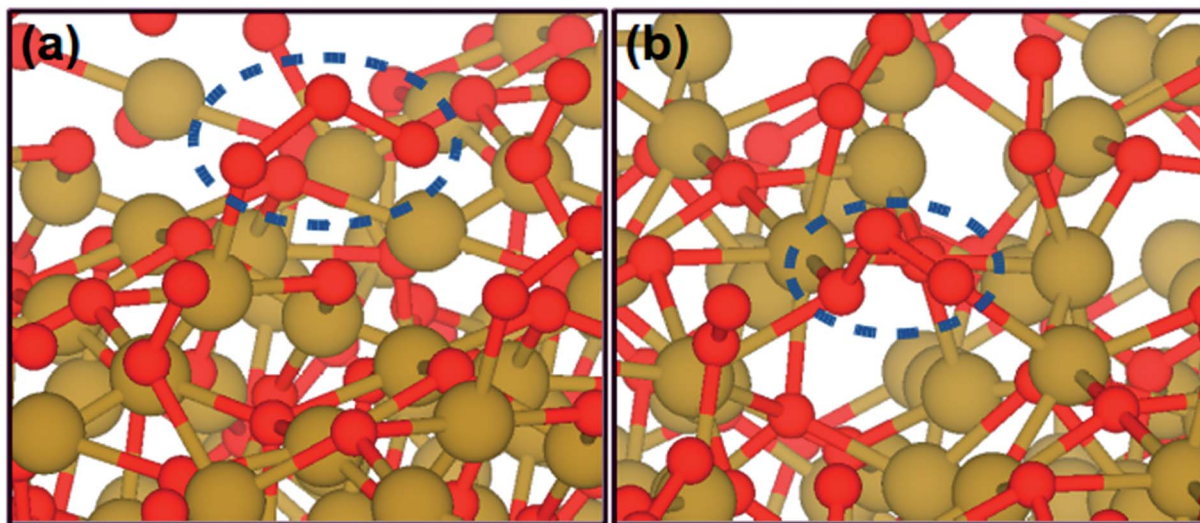


Fig. 7 Top views of  $\text{O}_3$  species over a (001)  $\text{Cu}_2\text{O}$  surface: (a) a chemisorbed  $\text{O}_3$  protrusion (dashed line) with one Cu–O bond. (b) An  $\text{O}_3$ -related bridge (dashed line) between surface Cu atoms. The  $\text{O}_3$  moiety forms three O–Cu bonds. (Cu: brown, O: red spheres).





## Conclusions

In summary, cuprous oxide nanocubes were successfully synthesized by a solution-based method at room temperature. When the Cu<sub>2</sub>O nanocubes were incorporated as the sensing element on IDEs, the resulting sensing element showed excellent sensitivity to O<sub>3</sub> in the range of 10 to 820 ppb, with fast response and recovery times. Importantly, the self-powered sensing elements of Cu<sub>2</sub>O nanocubes exhibited a high response (28%) at relatively low ozone concentration (10 ppb) with response and recovery times of 30 and 18 s, respectively. Finally, the p-type nature of the Cu<sub>2</sub>O nanocubes was confirmed, as the device current was found to increase in the presence of a strongly oxidizing gas, such as ozone.

## Conflicts of interest

There are no conflicts to declare.

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