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## Platinum-free photoelectrochromic devices working with copper-based electrolytes for ultrastable smart windows†

Luca Lavagna,<sup>ab</sup> George Syrokostas,<sup>c</sup> Lucia Fagiolari,<sup>a</sup> Julia Amici,<sup>ab</sup> Carlotta Francia,<sup>ab</sup> Silvia Bodoardo,<sup>ab</sup> George Leftheriotis<sup>\*c</sup> and Federico Bella<sup>ab</sup>

Photoelectrochromic systems are devices designed for large-scale manufacturing of smart windows, capable of changing their transmittance according to external environmental conditions. This communication proposes the replacement of the two most critical photoelectrochemical device components studied so far, namely the counter electrode and the redox mediator. Regarding the first, graphene nanoplatelets are used to replace platinum, maintaining both its optical and electrocatalytic properties, and at the same time reducing the device cost. Secondly, a copper-based redox pair was chosen to solve the corrosion problems typically encountered with the iodine-based mediator. The combination of the above components led to devices with high performance (coloration speeds in the order of seconds, with a maximum contrast ratio of 10.4 : 1), as well as the achievement of a long-term stability record (over 400 days) for these photoelectrochromic systems.

### Introduction

Smart windows, as particular glazing components able to tune the quantity of light and heat transmitted in a building, thus alleviating air conditioning costs, are becoming indispensable when designing new urban or industrial areas.<sup>1–6</sup> To this purpose, photoelectrochromic devices (PECs), as a commingling of a dye-sensitized solar cell (DSSC) and an electrochromic device (ECD), represent one of the most reliable technologies for smart windows development.<sup>7–9</sup> They are generally fabricated placing a dye-covered nanostructured TiO<sub>2</sub> layer over a porous WO<sub>3</sub> anode deposited onto a conductive glass, while

the counter electrode (CE) is a thin platinum film and the electrolyte is a I<sup>−</sup>/I<sub>3</sub><sup>−</sup> solution containing lithium salts as additives.<sup>10,11</sup> The working principle of this device is based on sunlight absorption by chemisorbed dye molecules, followed by electron injection into the conduction band of TiO<sub>2</sub>, electron diffusion into WO<sub>3</sub> and simultaneous intercalation of Li<sup>+</sup> in the latter.<sup>12,13</sup> As a result, WO<sub>3</sub> switches from transparent to blue color and, if the irradiance is attenuated or the device is short-circuited, the transparency is reversibly regenerated.<sup>14</sup>

Although the scientific community has worked hard to develop new materials for PECs in the last 10 years,<sup>15,16</sup> also including the integration with perovskite-based photovoltaics<sup>17</sup> and large-area deposition techniques,<sup>18</sup> two major limitations are still present and constitute a relevant obstacle for the scale-up of this technology. The first is related to the use of critical raw materials or expensive elements, such as platinum (used as a PEC and ECD CE)<sup>19</sup> or cobalt (used for high voltage DSSC electrolytes).<sup>20</sup> The second limitation concerns the long-term stability, mainly affected by the evaporation of the organic solvents-based liquid electrolyte and the degradation of photosensitive components present in the cell (*e.g.*, molecular sensitizer, iodine-based redox shuttle, *etc.*).<sup>21–23</sup>

With the aim of designing efficient, stable and cobalt/platinum-free PECs, we identified graphene nanoplatelets (GNPs) and copper complexes as key components, both of them never explored before in this field. GNPs are raising a huge interest in the materials science field due to their superior mechanical, thermal and barrier properties, ascribed to their peculiar high aspect ratio and plate-like shape.<sup>24–26</sup> In PECs, GNPs could represent a winning choice as platinum-free CEs, guaranteeing good electrical conductivity and electrocatalytic activity for the redox couple regeneration and electrons transport, as recently demonstrated in the DSSCs field.<sup>27,28</sup> As regards tetra-coordinated copper(I/II) complexes, they revolutionized the DSSCs field due to their unique characteristics as redox shuttles, *i.e.* high open-circuit voltage ( $V_{oc}$ , >1 V) and fast regeneration at low driving force, due to a minimized internal reorganization energy derived from the preservation of the

<sup>a</sup>Department of Applied Science and Technology, Politecnico di Torino, Corso Duca degli Abruzzi 24, 10129 Torino, Italy. E-mail: federico.bella@polito.it

<sup>b</sup>National Interuniversity Consortium of Material Science and Technology (INSTM), Via Giuseppe Giusti 9, 50121 Firenze, Italy

<sup>c</sup>Physics Department, Renewable Energy Lab, University of Patras, 26500 Rion, Greece. E-mail: gleftther@physics.upatras.gr

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complex geometry during the change of oxidation state.<sup>29–31</sup> In this study, we demonstrate an extremely efficient PECVD exploiting GNPs as the cathodic material and  $\text{Cu}(\text{dmp})_2\text{TFSI}/\text{Cu}(\text{dmp})_2(\text{TFSI})_2$  as redox shuttle; a record long-term stability exceeding 400 days was achieved by jellifying the redox mediator in poly(ethylene oxide) (PEO), thus avoiding liquid electrolyte leakage and easing device fabrication at the same time.

## Results and discussion

PECVDs were prepared through three steps. First,  $\text{TiO}_2/\text{WO}_3$  mesoporous electrodes were fabricated, where a  $\text{WO}_3$  layer ( $\approx 75\%$  transmittance in the visible range, thickness between 400 and 600 nm, packing density calculated of 0.795) was partially covered with a  $\text{TiO}_2$  film (thickness = 8  $\mu\text{m}$ ). 3-[6-[4-[bis(2',4'-dihexyloxybiphenyl-4-yl)amino]-phenyl]-4,4-dihexyl-cyclopenta-[2,1-b:3,4-b'] dithiophene-2-yl]-2-cyanoacrylic acid (Y123 dye) was chosen as metal-free molecule to sensitize the  $\text{TiO}_2$  film. Second, bis-(2,9-dimethyl-1,10-phenanthroline)copper(i) bis(trifluoromethanesulfonyl)imide ( $\text{Cu}(\text{dmp})_2\text{TFSI}$ ) and bis-(2,9-dimethyl-1,10-phenanthroline)copper(ii) bis(bis(trifluoromethanesulfonyl)imide) ( $\text{Cu}(\text{dmp})_2(\text{TFSI})_2$ ) were chosen as reduced and oxidized species of the copper-based redox mediator, respectively. Their concentration in acetonitrile was fixed at 0.10 and 0.03 M, respectively, while 4-*tert*-butylpyridine 0.20 M and lithium bis(trifluoromethanesulfonyl)imide 0.20 M were used as additives. The so-obtained liquid electrolyte was jellified by PEO, added at a concentration equal to 45 wt%, and spread onto the  $\text{TiO}_2/\text{WO}_3$  electrode. Third, CE were prepared by slowly dispensing a dilute GNPs suspension onto conductive glasses, followed by a drying process. The dimension of the resulting devices was  $3.0 \times 4.0 \text{ cm}^2$ ; the two electrodes were arranged facing each other, slightly displaced along their longitudinal axis to preserve space for electrical contacts. A thermoplastic film with a thickness equal to 25  $\mu\text{m}$  was placed between the two electrodes in order to peripherally seal the PECVD.

GNPs successfully and homogeneously covered the conductive glass, remaining rather invisible at human eye and

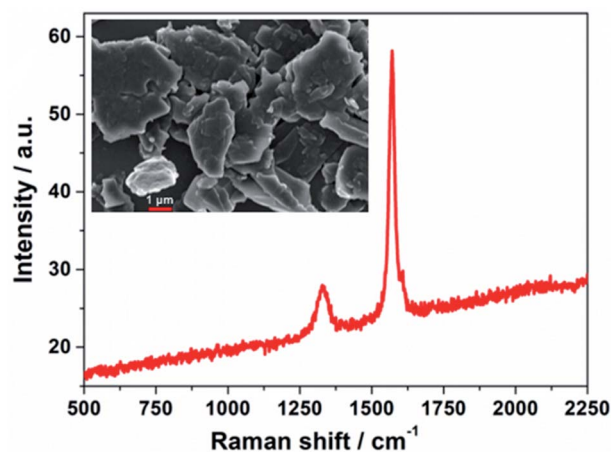


Fig. 1 Raman spectrum of GNPs used as CE material. Inset: FESEM micrograph of GNPs onto the PECVD conductive glass, just after deposition and solvent evaporation.

preserving the PECVD transmittance. The field emission scanning electron microscopy (FESEM) micrograph, shown in the inset of Fig. 1, depicts a GNPs morphology characterized by the presence of few single/multi-layer sheets and agglomerates of sheets, the dimensions of which are in good agreement with the data reported by the supplier.<sup>32</sup> The Raman spectrum (Fig. 1) presents two characteristic bands, namely the D-band at  $\approx 1300 \text{ cm}^{-1}$ , arising from the disorder-induced phonon mode ( $A_{1g}$ -band), and the G-band at  $\approx 1600 \text{ cm}^{-1}$ , assigned to the Raman-allowed phonon mode ( $E_{2g}$ -band). For graphitic materials, the ratio of the D and G band intensities ( $I_D/I_G$ ) decreases with the degree of order of the graphitic structure;<sup>33</sup> however, for very small in-plane correlation length ( $<20 \text{ \AA}$ ), corresponding



Fig. 2 Schematic diagram of the PECVD in: (A and B) open-circuit conditions under simulated sunlight (at  $t = 0$  and after 10 s); (C) short-circuit under dark conditions. The oxidation and reduction of the Y123 dye is not drawn so as not to complicate the readability of panels (A) and (C).

to small hexagonally ordered carbon clusters, the  $I_D/I_G$  ratio decreases again, reaching zero. After baseline correction and peak deconvolution, an average  $I_D/I_G = 0.54$  was calculated for our samples. Zeta potential measurements led to a value equal to  $-12.9 \pm 0.2$  mV, indicative of good stability for a colloidal system. As regards the hydrodynamic size, a value of  $2502 \pm 13$  nm was determined. The thermal stability of the GNPs after deposition is shown in Fig. S1†

After assembly, the coloration and bleaching processes were carried out by repeatedly illuminating the PECD in open-circuit conditions and keeping it short-circuited in the dark, respectively. The overall schematic diagram of the PECD is given in Fig. 2 for both open- and short-circuit conditions. The PECD showed a fast coloration step, *i.e.* its transmittance was halved after just 5 s of exposure under simulated sunlight, as shown in Fig. 3A. The bleaching process – with reference to Fig. 3B – was rather rapid as well, even during the first cycles when the devices typically have to reach equilibrium conditions (*e.g.*, when complete wettability of the  $\text{WO}_3$  electrode by the electrolyte is achieved).

The light response of the PECD in the visible range was studied under an illumination of  $1000 \text{ W m}^{-2}$ , and Fig. 3C shows the transmittance spectra under bleached and colored conditions, respectively. At the first cycle, the highest

transmittance value was 67.1% when bleached, while it decreased up to 5.5% at 530 nm when colored. The PECD was cycled 5 times per day from Monday to Friday for 400 days, *i.e.* the longest ever reported ageing test for PECDs – to the best of our knowledge (a detailed comparison is given in Table S1†); between measurements, the devices were kept in open-circuit conditions onto laboratory benches. After 400 days of activity, no significant degradation was detected for transmittance modulation, *i.e.* the final bleached/colored values were 62.8% and 6.0%, respectively, implying a robust cycle stability of the device and a very slight decrease of transmittance under bleached conditions, due to minor device fabrication defects (typical of lab-scale prototypes). Data shown in Fig. 3C refer to the best PECD over a batch of 5 prototypes; anyway, the chosen device well represents the overall trend, for which the average transmittance values at day 400 were  $61.1 \pm 1.7\%$  and  $5.7 \pm 0.3\%$  under bleached and colored conditions, respectively.

When assessing the performance of PECDs, the response time represents a key figure of merit, measured by adjusting the circuit open and closed for a fixed amount of time under continuous illumination. Fig. 3D shows that the response time for the PECD was around 12 s for the coloring process, while the bleaching step took  $\approx 11$  s. The experiment was repeated in

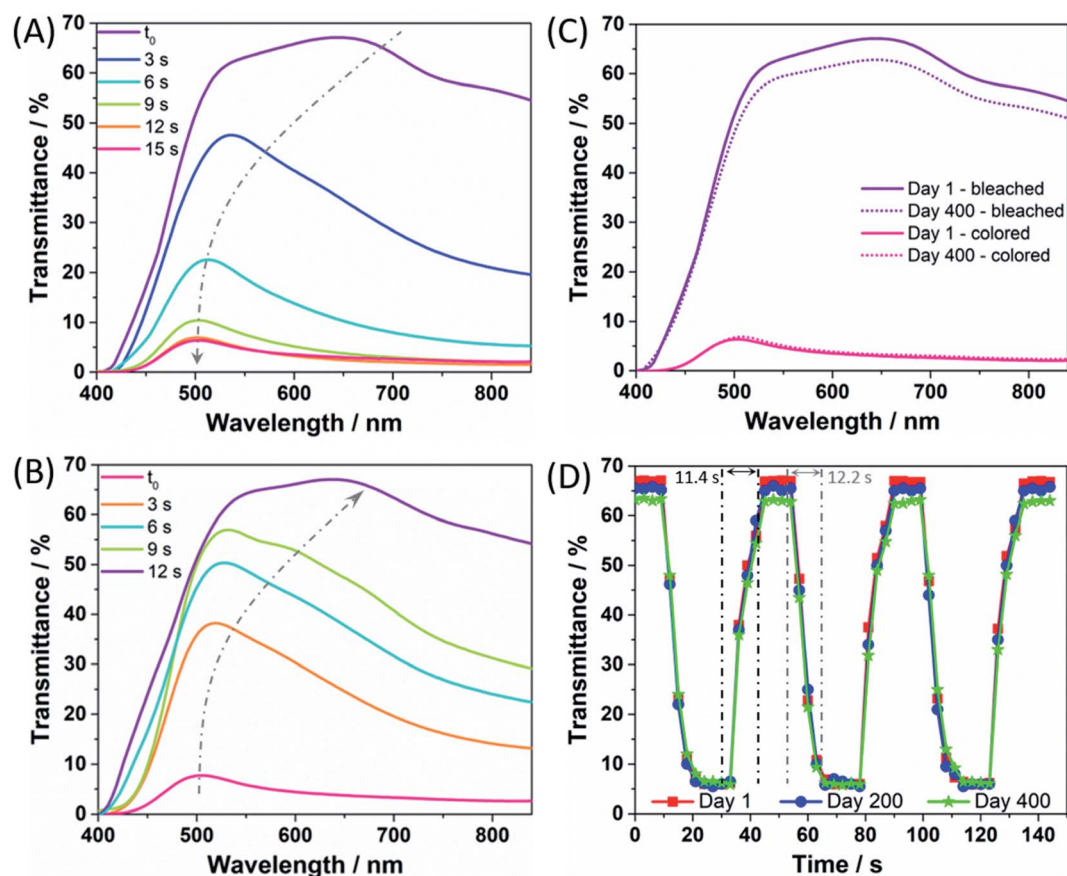


Fig. 3 Transmittance spectra for the PECD: (A) coloration process occurring under  $1000 \text{ W m}^{-2}$  (AM 1.5G) simulated sunlight and (B) bleaching process taking place under short-circuit and dark conditions. (C) Transmittance modulation of the PECD under bleached and colored conditions, comparing the spectra at day 1 and day 400 of the ageing test. (D) Transient transmittance response time for the PECD, comparing the traces at day 1, day 200 and day 400 of the ageing test.



Fig. 4 (A) PhCE of the PECD under 1 sun irradiation, comparing the values calculated at day 1, day 200 and day 400 of the ageing test. The inset shows pictures taken at day 400 under colored and bleached conditions, respectively. (B) Luminous transmittance vs. ageing time for the PECD. (C)  $J$ - $V$  curves of the DSSC portion of the PECD, measured under 1 sun irradiation (AM 1.5G) just after fabrication and at day 400 of the ageing test.

a long-term stability test and the response time profiles detected after 200 and 400 days are also shown in the plot. Overall, no significant performance degradation was found (in agreement with Fig. 3C), thus consolidating the excellent stability of the newly proposed copper- and GNPs-based redox shuttle and CE for PECDs application, respectively.

Photocoloration efficiency (PhCE), *i.e.* the change in the optical density of the PECD per unit light intensity applied per minute, was calculated during the coloration process and the results are shown in Fig. 4A. The highest value occurred during the first seconds, then PhCE values showed a gradual drop, overall indicating that the PECD underwent a rapid coloration process in a relative short exposure time. Indeed, the particular form of the PhCE curves is directly related to the coloration kinetics. As it can be seen in Fig. 3A, coloration progresses fast in the first few seconds of exposure. However, the rate of coloration decreases progressively as the  $\text{WO}_3$  film gets saturated with lithium ions, while the incoming energy density (denominator) continuously increases with time. Thus, PhCE also exhibits a continuous reduction with time. The evolution of PhCE values upon time was consistent with the long-term stability data previously discussed. The inset images represent the corresponding pictures taken under bleached and colored states, respectively.

The change in luminous transmittance of the PECD in its transparent, fully colored and bleached states *versus* days post fabrication is shown in Fig. 4B. A very limited gradual decrease in the color depth of the devices over time was detected, being – to the best of our knowledge – superior in terms of stability compared to all the previously published PECD ageing tests.<sup>7–11</sup> Besides minimal imperfections occurring during device assembly, the soft lowering in performance could be attributed to the progressive penetration of the copper-based electrolyte in the  $\text{WO}_3/\text{TiO}_2$  electrode. If – on the one hand – this improves the PECD working area, it also enhances recombination phenomena between the oxidized species of the redox shuttle with the electrons of the semiconductor conduction band, thus lowering the overall PECD performance. The quality of the GNPs-based CE was preserved upon the whole ageing test, as shown by the FESEM micrographs (Fig. S2†) recorded after cell disassembly.

The photocurrent density *vs.* photovoltage curves for the DSSC portion of the PECD were measured under simulated sunlight (AM 1.5G, 1 sun), and the resulting plots at days 1 and 400 are shown in Fig. 4C. An overall power conversion efficiency (PCE) of 2.43% was measured just after device fabrication, with  $V_{oc}$ , short-circuit current density ( $J_{sc}$ ) and fill factor (FF) equal to 0.86 V, 6.41  $\text{mA cm}^{-2}$  and 44%, respectively. Overall, the  $J$ - $V$  can be considered satisfactory, considering the fact that the  $\text{TiO}_2$ -based photoanodes were annealed at a low temperature, *i.e.* 120 °C, in order to preserve the quality of the underlying  $\text{WO}_3$  electrode (their FESEM images and X-ray diffraction spectra are shown in Fig. S3 and S4,† respectively). Also, the molar concentration of the reduced and oxidized species of the redox shuttle were kept at 0.10 and 0.03 M, respectively, *i.e.* lower than those commonly adopted in the DSSCs field,<sup>29–31</sup> in order to achieve high transparency of the PECD in the bleached state. The DSSC performance were kept rather stable ( $\approx 95\%$ ) upon

ageing, with PCE,  $V_{oc}$ ,  $J_{sc}$  and FF values equal to 2.13%, 0.83 V, 6.10 mA cm<sup>-2</sup> and 42%, respectively. Furthermore, given that DSSCs typically show very good operation under diffused sunlight, we carried out the transient transmittance response time study for the PECD under 0.5 sun. The plot shown in Fig. S5† clearly highlights that the coloration kinetics was slightly lowered when halving the sun simulator power output, but the PECD was able to reach the same colored state with a rather low delay, *i.e.* 9 s.

## Conclusions

PECDs manufactured with GNPs CEs and copper-based redox electrolytes have made it possible to overcome the use of platinum and iodine, respectively, thus solving two of the main drawbacks of this renewable energy-related technology, in terms of corrosion resistance and use of cheaper/abundant raw materials.

The fabricated PECDs exhibit an excellent performance combining a very fast coloration speed and an outstanding contrast ratio of 10.4 : 1. Furthermore, a record stability exceeding 400 days was reached, thus confirming the proper choice of the novel PECD components, included the jellification of the copper-based redox couple solution to avoid evaporation issues. The present findings pave the way to platinum-free smart windows suitable for energy saving applications, and the road towards the 70% transmittance target for building-integrated products is getting closer.

## Conflicts of interest

The authors declare no competing financial interest.

## References

- H. Zhang, J. Hao, X. Yu, J. Wu, J. Li, M. H. Li and J. Hu, *J. Mater. Chem. A*, 2020, **8**, 17800–17807.
- S. J. Lee, S. H. Lee, H. W. Kang, S. Nahm, B. H. Kim, H. Kim and S. H. Han, *Chem. Eng. J.*, 2021, **416**, 129028.
- Y. Zhou, X. Dong, Y. Mi, F. Fan, Q. Xu, H. Zhao, S. Wang and Y. Long, *J. Mater. Chem. A*, 2020, **8**, 10007–10025.
- A. K. Chowdhary and D. Sikdar, *Sol. Energy Mater. Sol. Cells*, 2021, **222**, 110921.
- S. Huang, Q. Zhang, F. Yang, D. T. Gangadharan, P. Li, F. Ren, B. Sun and D. Ma, *J. Mater. Chem. A*, 2020, **8**, 8620–8628.
- S. W. Oh, S. M. Nam, S. H. Kim, T. H. Yoon and W. S. Kim, *ACS Appl. Mater. Interfaces*, 2021, **13**, 5028–5033.
- A. Cannavale, P. Cossari, G. E. Eperon, S. Colella, F. Fiorito, G. Gigli, H. J. Snaith and A. Listorti, *Energy Environ. Sci.*, 2016, **9**, 2682–2719.
- A. Dokouzis, F. Bella, K. Theodosiou, C. Gerbaldi and G. Leftheriotis, *Mater. Today Energy*, 2020, **15**, 100365.
- A. Dokouzis, D. Zoi and G. Leftheriotis, *Materials*, 2020, **13**, 2565.
- A. Cannavale, F. Martellotta, F. Fiorito and U. Ayr, *Energies*, 2020, **13**, 1929.
- Z. Tong, Y. Tian, H. Zhang, X. Li, J. Ji, H. Qu, N. Li, J. Zhao and Y. Li, *Sci. China: Chem.*, 2017, **60**, 13–37.
- Z. Wang, H. C. Chiu, A. Paoletta, K. Zaghbi and G. P. Demopoulos, *ChemSusChem*, 2019, **12**, 2220–2230.
- C. Costa, I. Mesquita, L. Andrade and A. Mendes, *Electrochim. Acta*, 2016, **219**, 99–106.
- A. Kolay, N. T. Z. Potts, K. Sardar, E. A. Gibson and M. Deepa, *Sustainable Energy Fuels*, 2019, **3**, 514–528.
- E. Pulli, E. Rozzi and F. Bella, *Energy Convers. Manage.*, 2020, **219**, 112982.
- F. G. Cai, D. Zhou, Q. Q. Xiong, J. H. Zhang, X. L. Wang, C. D. Gu and J. P. Tu, *Sol. Energy Mater. Sol. Cells*, 2013, **117**, 231–238.
- X. Xia, Z. Ku, D. Zhou, Y. Zhong, Y. Zhang, Y. Wang, M. J. Huang, J. Tu and H. J. Fan, *Mater. Horiz.*, 2016, **3**, 588–595.
- G. Cai, P. Darmawan, X. Cheng and P. S. Lee, *Adv. Energy Mater.*, 2017, **7**, 1602598.
- S. Yun, P. D. Lund and A. Hinsch, *Energy Environ. Sci.*, 2015, **8**, 3495–3514.
- N. Mariotti, M. Bonomo, L. Fagiolari, N. Barbero, C. Gerbaldi, F. Bella and C. Barolo, *Green Chem.*, 2020, **22**, 7168–7218.
- F. Bella, A. Lamberti, A. Sacco, S. Bianco, A. Chiodoni and R. Bongiovanni, *J. Membr. Sci.*, 2014, **470**, 125–131.
- M. Gerosa, A. Sacco, A. Scalia, F. Bella, A. Chiodoni, M. Quaglio, E. Tresso and S. Bianco, *IEEE J. Photovoltaics*, 2016, **6**, 498–505.
- R. Shanti, F. Bella, Y. S. Salim, S. Y. Chee, S. Ramesh and K. Ramesh, *Mater. Des.*, 2016, **108**, 560–569.
- M. Bahiraei and N. Mazaheri, *Energy*, 2021, **218**, 119395.
- A. D. Pendergast, Z. Deng, F. Maroun, C. Renault and J. E. Dick, *ACS Nano*, 2021, **15**, 1250–1258.
- R. Moreno Araújo Pinheiro Lima and H. P. de Oliveira, *J. Energy Storage*, 2020, **28**, 101284.
- D. H. Kweon and J. B. Baek, *Adv. Mater.*, 2019, **31**, 1804440.
- C. K. Kim, H. M. Kim, M. Aftabuzzaman, I. Y. Jeon, S. H. Kang, Y. K. Eom, J. B. Baek and H. K. Kim, *Mater. Today Energy*, 2018, **9**, 67–73.
- Y. Saygili, M. Stojanovic, H. Michaels, J. Tjepelt, J. Teuscher, A. Massaro, M. Pavone, F. Giordano, S. M. Zakeeruddin, G. Boschloo, J. E. Moser, M. Grätzel, A. B. Muñoz-García, A. Hagfeldt and M. Freitag, *ACS Appl. Energy Mater.*, 2018, **1**, 4950–4962.
- Y. Wang and T. W. Hamann, *Chem. Commun.*, 2018, **54**, 12361–12364.
- Y. Saygili, M. Söderberg, N. Pellet, F. Giordano, Y. Cao, A. B. Muñoz-García, S. M. Zakeeruddin, N. Vlachopoulos, M. Pavone, G. Boschloo, L. Kavan, J. E. Moser, M. Grätzel, A. Hagfeldt and M. Freitag, *J. Am. Chem. Soc.*, 2016, **138**, 15087–15096.
- Graphene Nanoplatelets by Cheaptubes, accessed May 2021, <https://www.cheaptubes.com/product-category/graphene-nanoplatelets>.
- M. S. Dresselhaus, A. Jorio and R. Saito, *Annu. Rev. Condens. Matter Phys.*, 2010, **1**, 89–108.