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Enhancing perovskite solar cell efficiency: ZnO–WO₃ as an electron transport layer to minimize recombination losses

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Tungsten trioxide (WO₃), with strong electron affinity and recombination suppression, serves as an effective electron transport layer (ETL). Incorporating zinc oxide (ZnO) enhances its conductivity, forming a ZnO–WO₃ composite with improved charge extraction and energy level alignment. The novelty of this study is to introduce ZnO–WO₃ as an interlayer ETL in CsPbI₂Br₂-based perovskite solar cells, enabling superior device performance and stability. Both WO₃ and ZnO–WO₃ films were synthesized *via* sol–gel spin coating. X-ray diffraction (XRD) confirmed the monoclinic phase for both films, with ZnO–WO₃ exhibiting a larger crystallite size (67.7 nm) and lower dislocation density (2.18×10^{14} lines per m). Raman spectroscopy revealed additional ZnO vibrational modes, indicating lattice reinforcement and enhanced structural integrity. Scanning electron microscopy (SEM) shows that ZnO–WO₃ films have larger, more uniform grains and smoother morphology than WO₃, indicating improved film quality. UV-vis analysis showed a redshift and reduced bandgap (2.74 eV), while PL spectra indicated lower defect-related recombination. Time-resolved photoluminescence (TRPL) shows reduced average decay time for ZnO–WO₃, indicating faster carrier dynamics. Devices with ZnO–WO₃ achieved a power conversion efficiency of 12.87% due to reduced charge transfer resistance (21 Ω) and higher recombination resistance (4605 Ω), as confirmed by electrochemical Impedance Spectroscopy (EIS). External Quantum Efficiency (EQE) of 95% further demonstrated enhanced charge collection, establishing ZnO–WO₃ as a promising ETL for high-efficiency PSCs.

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1. Introduction

One of the most important issues of the modern era is the global energy crisis, which is being fueled by environmental concerns, growing energy demands, and the depletion of fossil fuel supplies.¹ Carbon emissions have surged as a result of the dependence on non-renewable energy sources, causing climate change and environmental deterioration.² The development of renewable and sustainable energy technology is essential to addressing these issues. Because of its abundance, sustainability, and potential for high energy conversion efficiency, solar energy has become an attractive option.³ Perovskite solar cells (PSCs) in particular have drawn a lot of attention because of their customizable optoelectronic characteristics, inexpensive fabrication costs, and high power conversion efficiency.^{4,5} The Russian mineralogist Lev A. Perovski identified several minerals, which included the CaTiO₃, PbTiO₃, SrTiO₃, and

BiFeO₃, by suggested the name of perovskite.⁶ In this particular perovskite family, the structure that is most often seen is ABX₃, where X is an anion (for example, oxygen, halogens, or alkali metals), A is a monovalent cation, and B is a divalent cation. During the formation of the perovskite crystal structure, the B cation is arranged in an octahedral coordination, and the [BX₆]^{n−} octahedra come together to create a network that shares corners with the A cation. The face centers of this structure are occupied by X anion, the body centers are occupied by B, and the cube corners are occupied by A cations and arranged in a cubic structure.^{6,7}

PSCs depend on the electron transport layer (ETL) for efficient electron extraction and transport as well as for suppressing charge recombination at the interface. TiO₂, SnO₂, ZnO, and WO₃ are among the many inorganic materials that have been studied as ETLs in PSCs due to their excellent electron mobility, stability, and appropriate band alignment with the perovskite absorber layer, respectively.⁸ TiO₂ has been employed extensively among these materials, but additional efficiency gains are hampered by its drawbacks, which include poor electrical conductivity and photocatalytic instability under UV irradiation.^{9,10} Although ZnO and SnO₂ have better electron mobility, they frequently have interface flaws that impair device performance. Although WO₃ has gained attention as a possible ETL

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candidate due to its high stability and adjustable band structure, its undesirable band alignment and comparatively low conductivity require adjustments for the best PSC performance.^{11,12}

The inorganic PSCs CsPbI₃, CsPbBr₃, and CsPbIBr₂ can be classified based on their perovskite composition.^{13,14} In contrast to CsPbI₃, which has phase instability, and CsPbBr₃, which has a greater bandgap restricting its light absorption, CsPbIBr₂ has demonstrated superior stability under ambient circumstances.¹⁵ With a balanced bandgap (~2.0 eV), CsPbIBr₂ provides reasonable light absorption capacity together with improved thermal and environmental stability.¹⁵ However, charge recombination and inadequate energy band alignment remain problems for CsPbIBr₂-based PSCs, preventing further PCE advancements despite these benefits. These restrictions can be successfully overcome by an optimized ETL through increased electron extraction efficiency, less interfacial defects, and improved charge transfer.¹⁶

The high electron affinity, broad bandgap, and exceptional chemical stability of WO₃ material, suggested the studied as an ETL.¹⁷ However, its performance in PSCs is limited by its poorer electrical conductivity and less-than-ideal band alignment with CsPbIBr₂.¹⁸ To get around these limitations, doping techniques like ZnO inclusion have been investigated. Because of its favorable conduction band alignment and strong electron mobility, ZnO can improve WO₃ charge transport characteristics.¹⁹ A clear and homogeneous electron transport layer with fewer trap states is produced using low-temperature sol-gel processed ZnO ETLs, improving charge transport and achieving a PCE of 8.77% reported by Mahmud *et al.*²⁰ Similar to this, ZnO ETLs manufactured in aqueous solution have a low work function and great transparency. Zhou *et al.* modified them with urea to promote perovskite crystal development, and they achieved a PCE of 14.6%.²¹ According to Song *et al.*, low-temperature metal oxide ETLs, such as WO₃, have outstanding optoelectronic qualities and adaptability for effective planar PSCs.²² You *et al.* found that a TiO₂/WO₃ bilayer ETL surpasses 20% efficiency in planar PSCs by efficiently minimizing energy loss and carrier recombination at interfaces.²³ The composite ZnO-WO₃ ETL enhances conductivity, lowers defect density, and guarantees improved energy level alignment with the perovskite absorber by integrating ZnO into WO₃. By boosting charge extraction and transport, this improvement lowers recombination losses and boosts PSC performance as a whole.²⁴

This study presents a novel approach to the synthesis of ZnO-WO₃ film that is utilized as an ETL with improved charge extraction and energy level alignment at the TiO₂/CsPbIBr₂ interface. Both WO₃ and ZnO-WO₃ films were synthesized *via* sol-gel spin coating. The prepared ETL film was analyzed through characterization such as XRD, Raman analysis, UV-vis spectroscopy, and PL. However, the produced PSCs were assessed by JV, EIS and EQE characterisation. It was found that ZnO-WO₃-based devices significantly outperformed pure WO₃ in terms of PCE. EIS confirms the higher recombination resistance and decreased charge transfer resistance, which further validates ZnO-WO₃'s improved performance. Measurements of

EQE show enhanced charge collection, highlighting the designed ETL's efficacy. These results open the door for more developments in inorganic perovskite photovoltaics by demonstrating ZnO-WO₃'s promise as a strong ETL material for stable, high-efficiency PSCs.

2. Experimental details

2.1. Materials required

Every chemical employed in this investigation was analytical grade and didn't need any further purification. Sigma-Aldrich was the supplier of tungsten(vi) chloride (WCl₆, 99.9%), zinc acetate dihydrate (Zn(CH₃COO)₂·2H₂O, 99.99%), ethanol (≥99.8%), acetic acid (glacial, 99.7%), cesium iodide (CsI, 99.999%), lead bromide (PbBr₂, 99.999%), *N, N*-dimethylformamide (DMF, anhydrous, 99.8%), dimethyl sulfoxide (DMSO, anhydrous, 99.9%), 4-*tert*-butylpyridine (tBP, 96%), lithium bis(trifluoromethanesulfonyl)imide (Li-TFSI, 99.95%), Spiro-OMeTAD (≥99%), and chlorobenzene (anhydrous, 99.8%). Glass substrates doped with fluorine (FTO) and having a sheet resistance of around 14 Ω per sq were acquired from Xinyan Technology Co. in China. Deionized (DI) water and anhydrous solvents were used to make each solution.

2.2. Synthesis of pure WO₃ and ZnO-WO₃ films

Sol-gel spin-coating was used,^{25,26} to create WO₃ and ZnO-WO₃ thin films. 198.7 mg of WCl₆ (0.5 mmol) was mixed in 10 mL of ethanol and continuously agitated for an hour at room temperature to create the WO₃ precursor solution. 200 μL of glacial acetic acid was added dropwise while stirring in order to stabilize the sol and regulate the hydrolysis.²⁵ To create a 10 mol% ZnO in WO₃ precursor (Zn : W = 1 : 9), 27.5 mg of zinc acetate dihydrate (0.125 mmol) was first dissolved in 2 mL of ethanol and then combined with the WO₃ solution for the ZnO-WO₃ composite.²⁷ To guarantee homogeneity, the resultant solution was agitated for an extra hour. Prior to deposition, both solutions were passed through a 0.45 μm PTFE syringe filter. Glass substrates coated with FTO were cleaned, sonicated for 15 minutes each in detergent, deionized water, acetone, and isopropanol, and then dried under nitrogen. The cleaned substrates were spin-coated with the filtered precursor solutions for 40 seconds at 3000 rpm. To guarantee phase formation and excellent crystallinity, the films were annealed at 500 °C for two hours in ambient air after being dried for ten minutes at 100 °C.^{25,27}

2.3. Devices fabrication

Three planar PSCs architectures were fabricated with the following device structures: FTO/TiO₂/CsPbIBr₂/Spiro-OMeTAD/Au, FTO/TiO₂/WO₃/CsPbIBr₂/Spiro-OMeTAD/Au, and FTO/TiO₂/ZnO-WO₃/CsPbIBr₂/Spiro-OMeTAD/Au. To produce a compact TiO₂ coating, a 0.15 M solution of titanium diisopropoxide bis(acetylacetonate) in 1-butanol was first spin-coated for 30 seconds at 2000 rpm and then annealed for 30 minutes at 500 °C.²⁸ Using the previously mentioned spin settings, an extra WO₃ or ZnO-WO₃ layer was deposited on top



of TiO₂ for the modified devices. 0.3 M CsI and 0.3 M PbBr₂ were dissolved in a 1 : 1 v/v mixture of DMF and DMSO to create the perovskite layer CsPbIBr₂. The solution was spin-coated in two stages: 10 seconds at 1000 rpm and 30 seconds at 4000 rpm. As an anti-solvent, 200 μ L of chlorobenzene was dripped in the last 10 seconds of the second phase.²⁸ To guarantee phase purity, the resultant perovskite film was annealed for ten minutes at 280 $^{\circ}$ C. By spin-coating a Spiro-OMeTAD solution, which was made by dissolving 72.3 mg of Spiro-OMeTAD in 1 mL of chlorobenzene and adding 17.5 μ L of Li-TFSI solution (520 mg mL⁻¹ in acetonitrile) and 28.8 μ L of 4-*tert*-butylpyridine, the hole transport layer was created. For 30 seconds, this HTL solution was spin-coated at 4000 rpm. Then, using a shadow mask and a high vacuum ($\sim 1 \times 10^{-6}$ Torr), a gold electrode (~ 80 nm thick) was thermally evaporated with a controlled deposition rate of 0.2 $\text{\AA} \text{ s}^{-1}$ for the first 30 nm and 1 $\text{\AA} \text{ s}^{-1}$ for the remaining thickness.²⁸

2.4. Characterizations

Using X-ray diffraction (XRD) recorded on a Bruker D8 Advance diffractometer equipped with Cu K α radiation ($\lambda = 1.5406 \text{ \AA}$) and operating at 40 kV and 40 mA with a step size of 0.02 $^{\circ}$ in the 2θ range of 10 $^{\circ}$ to 60 $^{\circ}$, the structural characteristics of WO₃ and ZnO-WO₃ films were described. Using a 532 nm excitation laser and a resolution of 1 cm⁻¹, Raman spectroscopy was performed using a Renishaw InVia Raman microscope. Field-emission scanning electron microscopy (FE-SEM) was used to examine the surface morphology using an FEI Nova NanoSEM 450 running at 5 kV. A Shimadzu UV-2600 spectrophotometer was used to record UV-visible absorption spectra in the 300–800 nm wavelength range. An Edinburgh Instruments FLS1000 spectrometer was used to gather steady-state photoluminescence (PL) and time-resolved PL (TRPL) spectra. A triple-exponential decay model was used to fit TRPL. A Keithley 2400 source meter and a Newport Oriel Sol3A solar simulator were used to test photovoltaic performance (J - V curves) under simulated AM 1.5 G light (100 mW cm⁻²). A Zahner Zennium electrochemical workstation was used to conduct electrochemical impedance spectroscopy (EIS) in the dark, operating in the frequency range of 1 MHz to 0.1 Hz. A PV Measurements QEX10 setup with a lock-in amplifier for signal augmentation was used to get external quantum efficiency (EQE) spectra.

3. Results discussion

3.1. Structural analysis

Materials crystalline structure, phase composition, and crystallite size can all be ascertained using XRD.²⁹ The XRD patterns of pure WO₃ and ZnO-WO₃ films are shown in Fig. 1a, with discrete peaks that correspond to their crystal planes. The monoclinic structure of the pure WO₃ film with space group $P2_1/a$ (14), as well as its distinctive diffraction peaks at the (011), ($\bar{1}11$), (111), (020), (002), ($\bar{2}11$), (120), (112), (022), ($\bar{2}03$), and (013) planes, are confirmed according to the PDF#54-0508.^{30,31} The ZnO-WO₃ film retains the same monoclinic phase, but extra peaks that belong to ZnO show up at the (200) planes

according to PDF#65-2880.^{32,33} These ZnO peaks show that ZnO was successfully incorporated into the WO₃ matrix, creating a composite structure.³⁴ By decreasing recombination losses and enhancing electron transport, ZnO can improve the electrical characteristics of WO₃, which is advantageous for its function as an ETL in PSCs. The sharper and more intense diffraction peaks show that the ZnO-WO₃ film has significantly higher crystallinity than pure WO₃.³⁵ This implies greater structural ordering, which may result in PSCs with superior charge transport characteristics. The dislocation line density (ρ), crystallite size (D), and d -spacing was calculated with the help of the expressions given below:^{36–39}

$$D = \frac{0.9\lambda}{\beta \cos \theta} \quad (1)$$

$$\rho = \frac{1}{D^2} \quad (2)$$

$$d = \frac{n\lambda}{2 \sin \theta} \quad (3)$$

The Miller indices (hkl), the diffraction order parameter (n), the angle of diffraction (θ), the wavelength (λ , 1.542 \AA), and the FWHM (β) were all included in the above formulas. The Scherrer formula is used to determine the crystallite size, which comes out to be 43.1 nm for pure WO₃ and 67.7 nm for ZnO-WO₃.⁴⁰ Because ZnO-WO₃ has bigger crystallites, there are fewer grain borders, which reduces charge trapping and increases carrier mobility. Crystal defects are represented by the dislocation density (DLD), which drops from 5.38×10^{14} lines per m in WO₃ to 2.18×10^{14} lines per m in ZnO-WO₃. By lowering charge carrier scattering sites, this defect reduction enhances charge transport even further.⁴⁰ According to Bragg's law, the d -spacing values for WO₃ and ZnO-WO₃ are 3.52 \AA and 3.51 \AA , respectively, and show slight structural alterations upon ZnO insertion.⁴¹ ZnO-WO₃'s marginally smaller d -spacing indicates better atomic packing, which may result in more effective charge transport. ZnO-WO₃ is a more effective ETL material due to its increased crystallite size and decreased defect density, which speeds up electron extraction and transport in PSCs and eventually improves photovoltaic efficiency.⁴¹

An essential technique for examining vibrational modes and material structural integrity is Raman spectroscopy. As seen in the image, the Raman spectra of ZnO-WO₃ and pure WO₃ films exhibit unique peaks that correlate to their distinctive vibrational modes (Fig. 2).⁴² The monoclinic phase of the pure WO₃ film is confirmed by the presence of two main peaks: one at 800 cm⁻¹, which corresponds to the O-W-O stretching mode, and another at 715 cm⁻¹, which is linked to W-O bending vibrations.⁴² Additional peaks at 432 cm⁻¹, 493 cm⁻¹, and 604 cm⁻¹ that are ascribed to Zn-O vibrational modes appear upon ZnO integration, indicating ZnO existence in the composite structure.⁴³ Strong contact between the Zn and W atoms is indicated by these ZnO-related peaks, which results in lattice distortions that improve structural stability. Better crystallinity is suggested by the amplified peak at 806 cm⁻¹, which



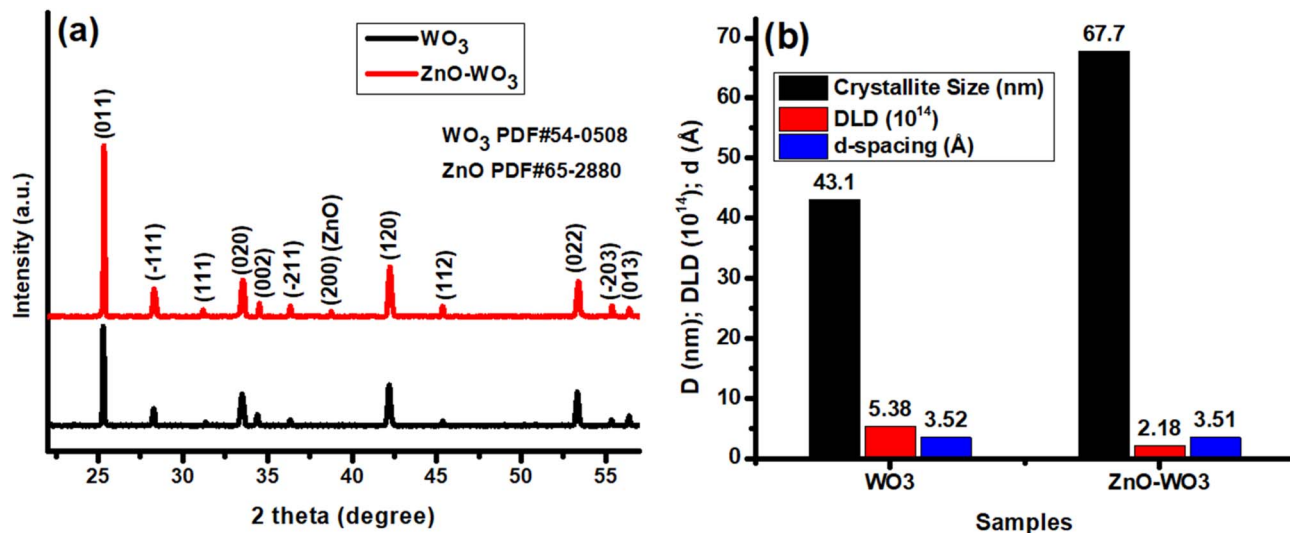


Fig. 1 (a) XRD patterns, (b) calculated values of crystallite size, dislocation line density, and *d*-spacing of WO₃ and ZnO-WO₃.

is advantageous for charge transport.⁴² ZnO is essential for electron transport layer (ETL) applications because it reduces defects and suppresses charge recombination, strengthening the lattice overall. ZnO-WO₃ is a good ETL candidate for perovskite solar cells (PSCs) because of the combined effect of WO₃ and ZnO, which guarantees greater stability, enhanced charge extraction, and superior structural integrity.⁴⁴

3.2. Optical analysis

The optical absorption properties of pure WO₃ and ZnO-WO₃ films are shown by their UV-vis absorption spectra (Fig. 3a). The ZnO-WO₃ film has a redshifted absorption edge at 403 nm, while the pure WO₃ film displays a crisp absorption edge at 383 nm.⁴⁵ The inclusion of ZnO, which alters the electronic structure of WO₃ by adding more energy levels close to the

conduction and valence bands, is responsible for this redshift.⁴⁶ For ETL applications, the redshift is advantageous because it increases light absorption, which boosts charge carrier production and transport efficiency in PSCs.⁴⁷ To determine the optical bandgap (E_g), the Tauc's relation was employed,^{48,49} as illustrated in Fig. 3b. The calculated bandgap values are 2.89 eV for pure WO₃ and 2.74 eV for ZnO-WO₃. Pure WO₃ has a larger bandgap because of the dominance of O 2p orbitals in the valence band and W⁶⁺ 5d orbitals in the conduction band.⁵⁰ Additional electronic interactions brought about by Zn²⁺ 4s and 3d orbitals in ZnO-WO₃ result in band hybridization and a little bandgap decrease.⁵¹ The addition of ZnO results in orbital hybridization and defect states, which marginally lowers the energy needed for electron excitation. ETLs in PSCs benefit from a lower E_g because it makes better charge injection and transfer possible, which increases device efficiency.⁵² Additionally, the film's extinction coefficient (K) and refractive index (n) (Fig. 3c) were examined.⁵³ WO₃ and ZnO-WO₃ have refractive index values of 2.38 and 2.43, respectively, and extinction coefficient values of 2.198 and 2.207. ZnO-WO₃ higher refractive index indicates greater light-matter interaction and photon confinement, which can improve optical absorption and charge transport characteristics. A higher absorption coefficient is shown by an increase in K , which lowers parasitic losses and enhances device performance overall.⁵⁴ Moreover, as illustrated in Fig. 3d, the imaginary dielectric constant (ϵ_i) and real dielectric constant (ϵ_r) were assessed.⁵³ WO₃ and ZnO-WO₃ have ϵ_r values of 0.83 and 1.03, respectively, and ϵ_i values of 10.46 and 10.72. The ZnO-WO₃ film's increased ϵ_r and ϵ_i show improved polarization and decreased electron scattering, which raises charge carrier mobility. More optical absorption is reflected by a greater ϵ_i , which promotes effective charge production. By increasing electron extraction, decreasing recombination losses, and boosting the general stability and effectiveness of PSCs, these characteristics make ZnO-WO₃ an excellent ETL choice.⁵⁵

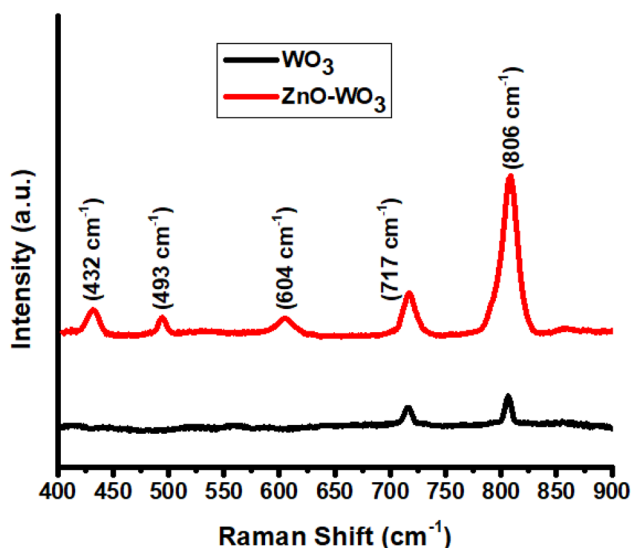


Fig. 2 Raman spectra of WO₃ and ZnO-WO₃.



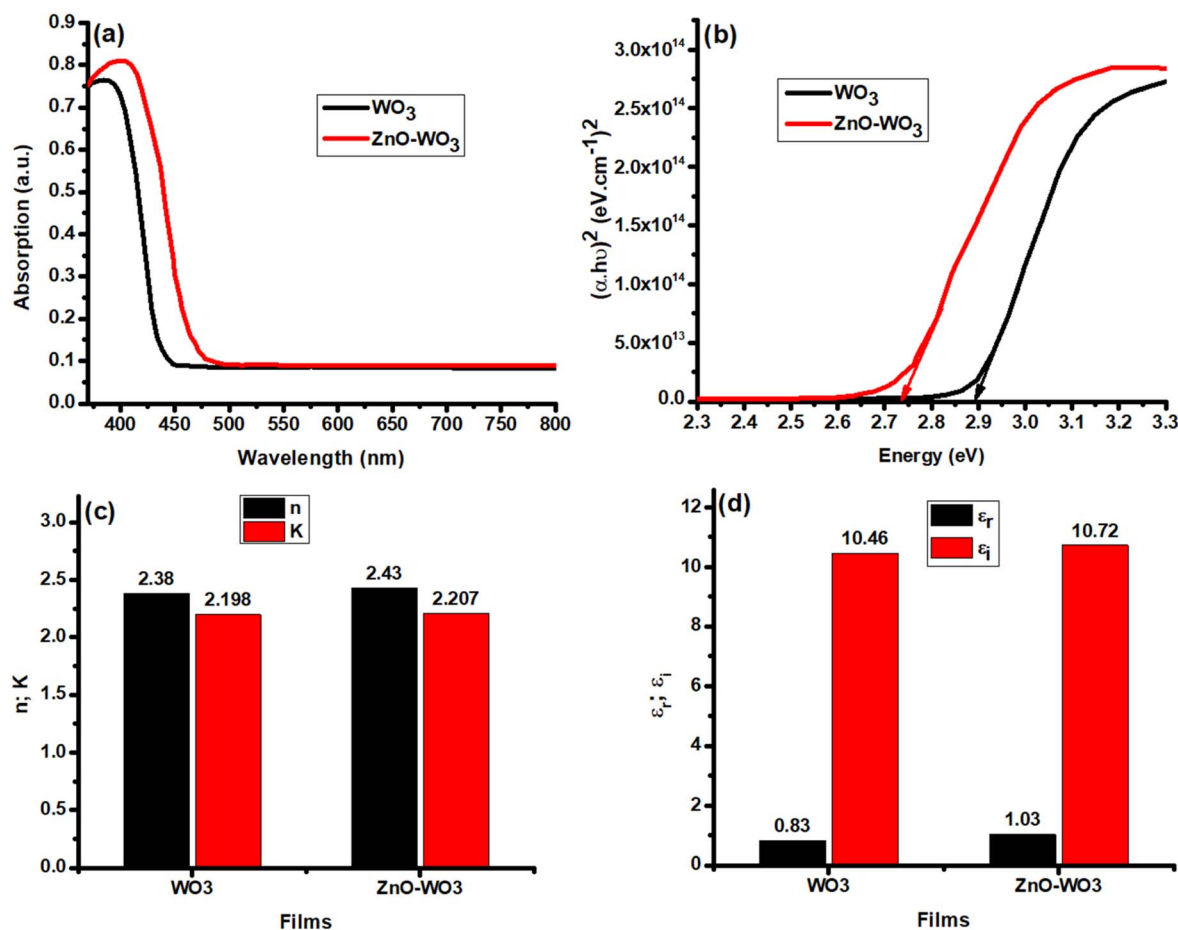


Fig. 3 (a) UV-vis absorption, (b) Tauc plots, (c) calculated values of refractive index and extinction coefficient, (d) calculated values of real and imaginary dielectric constant for WO_3 and ZnO-WO_3 .

The photoluminescence (PL) spectra of WO_3 and ZnO-WO_3 films were shown the excitation wavelength of 402 nm for ZnO-WO_3 and 382 nm for WO_3 . The observed optical transitions are confirmed by the good alignment of these excitation wavelengths with the UV-vis absorption measurements.⁵⁶ A redshift in the PL emission from the ZnO-WO_3 film indicates that the ZnO inclusion has changed the band structure.⁵⁷ This change results from Zn^{2+} doping-induced defect states that modify the kinetics of charge carrier recombination. Specifically in the higher-wavelength region, the ZnO-WO_3 film exhibits a notable decrease in defect-related emission intensity, indicating enhanced crystallinity and fewer non-radiative recombination sites. Defect peak suppression points to improved charge carrier transit and decreased trap-assisted recombination, both of which are essential for a successful ETL in PSCs.⁵⁷ By facilitating better electron extraction and transport, the redshift and decreased defect intensity minimize energy losses and raise the overall efficiency of perovskite solar cells.^{57,58} Average carrier lifetimes for WO_3 and ZnO-WO_3 ETLs were extracted by fitting the time-resolved photoluminescence (TRPL) data with a triple-exponential decay model (Fig. 4b). A remarkable average decay duration of around 119 ns was observed for WO_3 , suggesting considerable carrier trapping or defect-related recombination

suppression that might restrict effective charge extraction.⁵⁹ The ZnO-WO_3 composite, on the other hand, had a lower average decay time of 13.5 ns, indicating decreased trap-assisted recombination and quicker carrier dynamics.⁵⁹ The ZnO incorporation to WO_3 improved the ETL's capacity for charge transfer and extraction, which made it more appropriate for high-performance PSC applications. Superior device efficiency may result from this enhanced dynamic behavior, which may make it easier to pump electrons into the ETL from the perovskite absorber layer.⁵⁹

The surface morphology and grain size distribution of WO_3 and ZnO-WO_3 films, as determined by SEM and particle size analysis, are shown in Fig. 4c-f. The WO_3 film in Fig. 4c has smaller, agglomerated grains and a compact but irregular morphology, while Fig. 4d shows the matching particle size distribution with an average grain size of 18.3 μm .⁶⁰ On the other hand, the ZnO-WO_3 film in Fig. 4e exhibits bigger, clearly defined spherical aggregates together with a more consistent and smoother surface shape.⁶¹ Fig. 4f's size distribution shows a noticeably higher average particle size of 32.1 μm . For ETLs in PSCs, the addition of ZnO to WO_3 improves film uniformity and encourages grain development.⁶² Smoother carrier paths and less charge trapping and recombination are made possible by

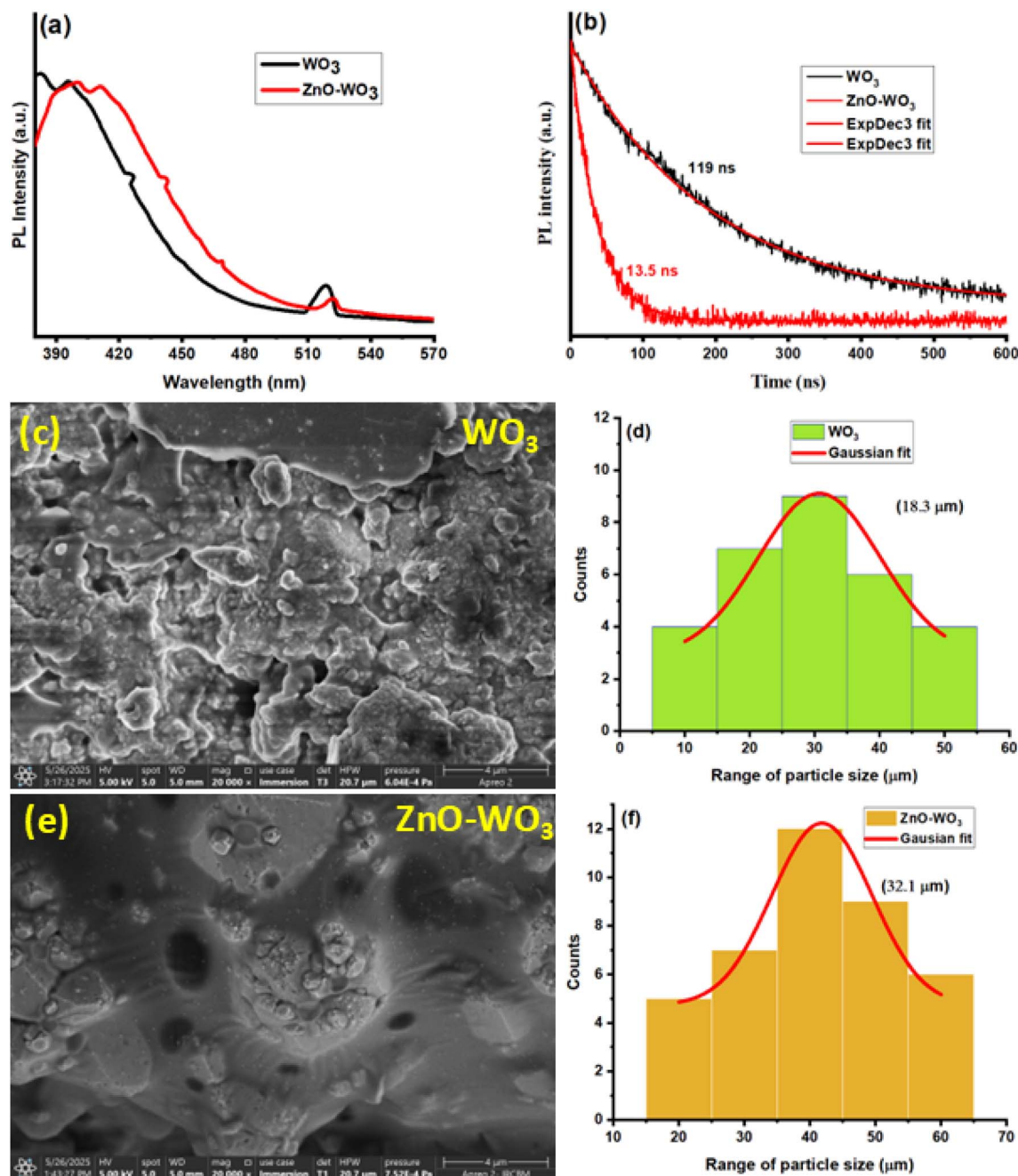


Fig. 4 (a) PL, (b) time resolved PL, (c–f) SEM images with corresponding particle size distributions of WO_3 and ZnO-WO_3 .

larger grain sizes and better morphology, which also lowers grain boundary density. According to these structural enhancements, the ZnO-WO_3 composite is a better ETL option than virgin WO_3 for improving PSC performance.^{62,63}

3.3. Photovoltaic performance

Three PSCs devices were constructed with the configurations (FTO/ TiO_2 /CsPbBr₂/Spiro-OMeTAD/Au), (FTO/ TiO_2 / WO_3 /CsPbBr₂/Spiro-OMeTAD/Au), and (FTO/ TiO_2 / ZnO-WO_3 /CsPbBr₂/Spiro-OMeTAD/Au) as shown in Fig. 5a to assess the efficiency of the produced films as electron transport layers

(ETLs). To modulate the conduction band offset (CBO) between TiO_2 and CsPbBr₂ and enable effective charge transfer, ZnO-WO_3 between TiO_2 and the perovskite layer is essential.⁶⁴ Electron-hole pairs are created when the perovskite absorber absorbs light. To promote a cascade-like electron transport process, electrons from the perovskite's conduction band (CB) are injected into ZnO-WO_3 CB and then into TiO_3 . The perovskite layer's electron-hole recombination is greatly decreased by this structured energy alignment, improving the efficiency of charge collecting.⁶⁴ The J - V curves and derived photovoltaic characteristics for three perovskite solar cells with various ETL



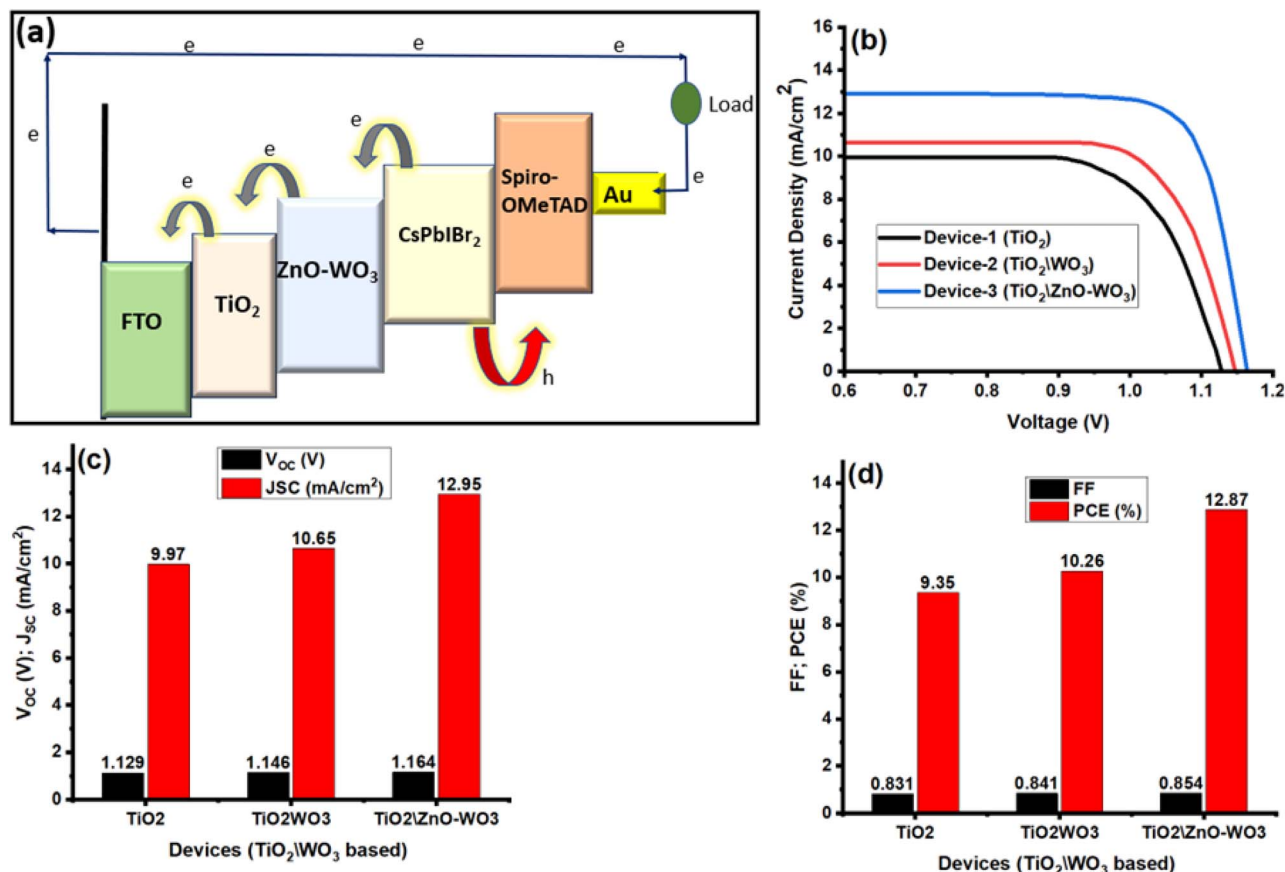


Fig. 5 (a) TiO₂/ZnO-WO₃ ETLs-based device (b) *J*-*V* measurement (c) calculated values of open-circuit voltage and short-circuit current density (d) calculated values of fill factor and power conversion efficiency for TiO₂, TiO₂/WO₃ and TiO₂/ZnO-WO₃ ETLs-based PSCs devices.

setups are shown in Fig. 5b-c: Device-1 (FTO/TiO₂/CsPbBr₂/Spiro-OMeTAD/Au), Device-2 (FTO/TiO₂/WO₃/CsPbBr₂/Spiro-OMeTAD/Au), and Device-3 (FTO/TiO₂/ZnO-WO₃/CsPbBr₂/Spiro-OMeTAD/Au). The following formulae are used to compute the efficiency (η) and fill factor (FF):⁶⁵

$$FF = \frac{P_{\max}}{J_{\text{sc}} \times V_{\text{oc}}} \quad (4)$$

$$\eta = \frac{V_{\text{oc}} \times J_{\text{sc}} \times FF}{P_{\text{in}}} \quad (5)$$

Device-1 displays a power conversion efficiency (PCE) of 9.35%, a fill factor (FF) of 0.831, a J_{sc} of 9.97 mA cm⁻², and a V_{oc} of 1.129 V. The addition of WO₃ between the perovskite and TiO₂ in Device-2 raises the PCE to 10.26%, the FF to 0.841, the J_{sc} to 10.65 mA cm⁻², and the V_{oc} to 1.146 V. This enhancement results from WO₃ advantageous conduction band edge alignment, which lowers energy losses and facilitates effective electron extraction from the perovskite layer to the ETL.⁶⁶ With a V_{oc} of 1.164 V, J_{sc} of 12.95 mA cm⁻², FF of 0.854, and PCE of 12.87%, Device-3 performs the best. As shown by longer carrier lifetimes in TR-PL measurements, the improvement in Device-3 results from the addition of ZnO to the WO₃ layer, which inhibits trap-assisted recombination and enhances carrier

transport. The TiO₂/ZnO-WO₃-based device performs noticeably better due to the combined effects of optimized energy level alignment by WO₃, less recombination losses, and quicker carrier dynamics by ZnO.^{23,66}

Electrochemical Impedance Spectroscopy (EIS) was conducted to analyze the charge transport and recombination processes within the fabricated perovskite solar cells (PSCs) employing WO₃ and ZnO-WO₃ as the electron transport layers (ETLs). The Nyquist plots for both devices are shown in Fig. 6a, where the charge transfer resistance (R_{ct}) is represented by the semicircle diameter. It is clear that the ZnO-WO₃-based device has a much smaller semicircle than the WO₃-based device, which suggests a lower resistance to charge transfer.⁶⁷ The retrieved parameters from EIS experiments are quantitatively shown in Fig. 6b, where the R_{ct} values for TiO₂/WO₃ and TiO₂/ZnO-WO₃ ETL-based devices are 79 Ω and 21 Ω , respectively. The ZnO-WO₃-based device's significant R_{ct} decrease points to improved electron injection and transport at the ETL/perovskite interface.⁶⁸ Furthermore, the WO₃-based device's recombination resistance (R_{rec}), which measures the device's capacity to inhibit charge carrier recombination, is found to be 4342 Ω , whereas the ZnO-WO₃-based device's R_{rec} rises to 4605 Ω . The ZnO-WO₃-based device's higher R_{rec} value validates the suppression of charge recombination, which enhances photovoltaic performance overall.⁶⁸

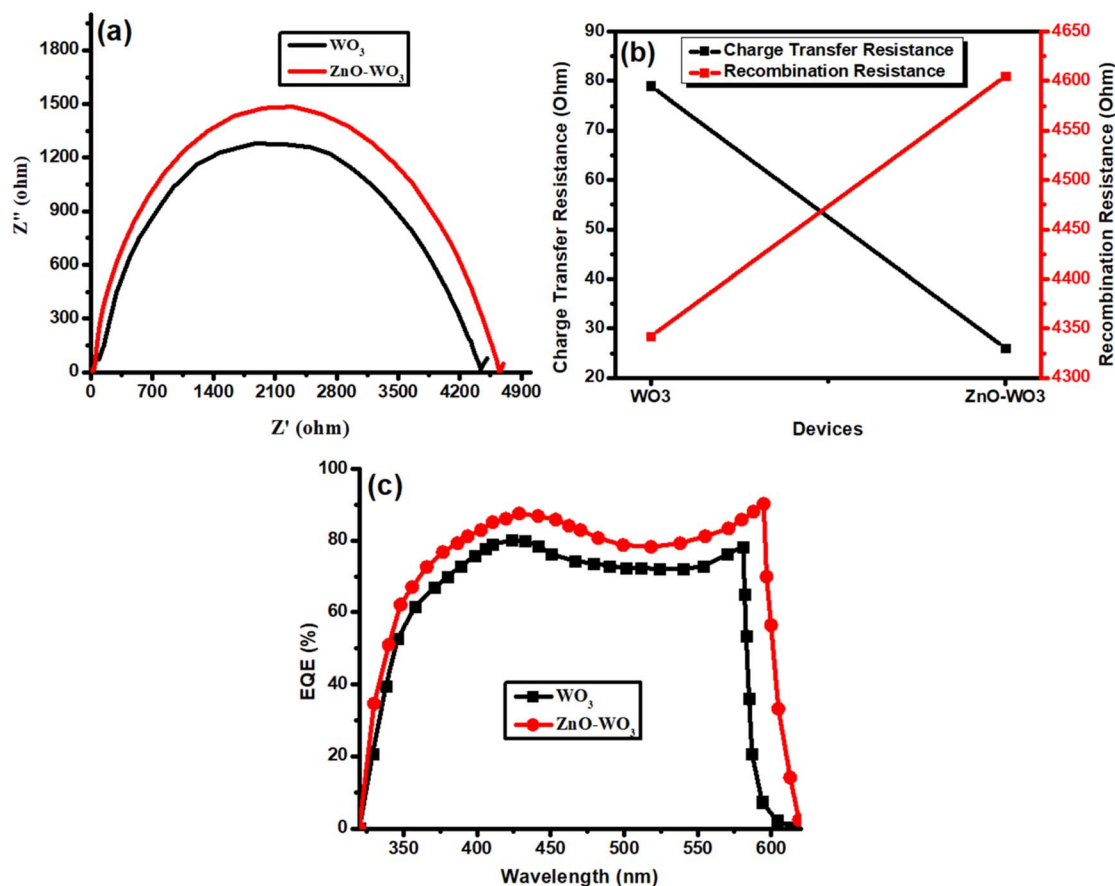


Fig. 6 (a) EIS spectra (b) calculated values of charge transfer and recombination resistances (c) EQE for TiO_2/WO_3 and $\text{TiO}_2/\text{ZnO-WO}_3$ ETLs based devices.

Fig. 6c shows the External Quantum Efficiency (EQE) spectra for both devices, which further confirms the ZnO-WO_3 -based ETL's better charge-collecting efficiency. In the wavelength range of 350 to 600 nm, the ZnO-WO_3 -based device's EQE response shows an overall improvement, with a peak efficiency of about 95%, as opposed to the WO_3 -based device's roughly 85%.⁶⁴ In line with the lower R_{ct} and higher R_{rec} seen in the EIS study, this improvement suggests a more effective extraction and transport of photogenerated charge carriers. By facilitating higher electron mobility, reducing energy barrier mismatches, and minimizing recombination losses, ZnO-WO_3 improves the power conversion efficiency (PCE) of PSCs, as confirmed by the increased EQE.⁶⁹

4. Conclusions

The incorporation of ZnO-WO_3 as an ETL between TiO_2 and the perovskite absorber effectively reduces recombination losses, leading to improved charge extraction and transport in PSCs. The sol-gel spin coating method was employed to synthesize both WO_3 and ZnO-WO_3 films. XRD analysis confirms the monoclinic phase of WO_3 . Raman spectroscopy revealed additional ZnO vibrational modes, indicating lattice reinforcement and enhanced structural integrity. Optical studies reveal a redshifted absorption edge and reduced bandgap, promoting efficient light absorption. SEM

shows that ZnO-WO_3 films have larger, more uniform grains and smoother morphology than WO_3 , indicating improved film quality. Time-resolved photoluminescence shows reduced average decay time for ZnO-WO_3 , indicating faster carrier dynamics. PSCs were fabricated with both ETLs, and devices with ZnO-WO_3 demonstrated superior performance, achieving a PCE of 12.87%. EIS analysis confirms reduced charge transfer resistance and improved recombination resistance, while EQE measurements show enhanced charge collection. These results highlight ZnO-WO_3 as a promising ETL for advancing high-efficiency PSCs in future photovoltaic applications.

Data availability

Data supporting this study are available from the corresponding author upon reasonable request.

Conflicts of interest

The authors have no conflict of interest.

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