RSC
Applied Interfaces

Accepted Manuscript

RSC
Applied Interfaces

GOLD
OPEN
ACCESS

ROYAL SOCIETY
OF CHEMISTRY

This is an Accepted Manuscript, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. We will replace this Accepted Manuscript with the edited and formatted Advance Article as soon as it is available.

You can find more information about Accepted Manuscripts in the Information for Authors.

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard Terms & Conditions and the Ethical guidelines still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this Accepted Manuscript or any consequences arising from the use of any information it contains.

ROYAL SOCIETY
OF CHEMISTRY

rsc.li/RSCAppIInter

1 Recent advances in interface engineering of bismuth-based materials for photocatalytic CO_2

2 reduction

3 Swetha S. M. Bhat[a]* and Girish Kumar S.[a]

4 [a]Department of Chemistry, RV College of Engineering, Mysuru Road, Bengaluru-560059,

5 India. E-mail: swethasm@rvce.edu.in

6 **Abstract**

7 Photocatalytic reduction of CO_2 into value added products is witnessed as a promising

8 technology to mitigate both global warming and energy crisis. Among the illustrious functional

9 semiconductors, bismuth-based materials are feasible for the CO_2 reduction reaction owing to

10 their convenient preparation, narrow band gap and the availability of hybridized energy states

11 in the conduction band. Furthermore, the inherent 2D layered structure of few Bi-materials

12 remains as an added advantage for various photocatalytic reactions. In this focused review,

13 interfacial engineering of Bi-based semiconductors achieved by coupling them with distinct

14 photocatalytic materials to form Type-I, Type-II, Z-scheme or S-scheme heterojunction are

15 discussed and their application towards CO_2 reduction reactions are emphasized. The further

16 advancements including the co-catalyst loading, defect engineering and designing with

17 hierarchical morphology from the prospect of improving the charge carrier separation and

18 structural stability are highlighted. The preparation methods and the mechanistic pathways for

19 the CO_2 reduction reaction are briefly summarized at the relevant places. Finally, the challenges

20 and scope of Bi-based materials to spotlight their applications in energy-environmental related

21 areas are presented.

22

23

24

25
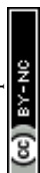
1.0 Introduction

Global energy consumption has skyrocketed in recent decade using nonrenewable fossil fuels, leading to significant global climate changes and imbalance in the ecosystem. As per the literature, CO_2 emission has substantially increased by ~40 % since the industrial revolution.[1, 2] Therefore, the development of efficient method to convert CO_2 to value added products have drawn the attention as they can conveniently address the environmental issues and the energy demands.[3] Numerous technologies such as chemical reforming, biological, electrochemical, photocatalytic, and photoelectrocatalytic processes are extensively investigated over the years to address these issues.[4-9] Among them, photocatalytic CO_2 reduction is found to be a sustainable solution for the removal of atmospheric CO_2 and transform them to selective gaseous (methane, ethane) and liquid phase products (formate, methanol, ethanol).[10-12] Compared to other approaches, this technique benefits from superior product selectivity, economic viability and environmentally benign.[13] The linear CO_2 with sp^2 hybridization in C=O bonds require more dissociation energy compared to C-C and C-H bond and adequate amount of energy is required to activate CO_2 molecules.[14, 15] The primary step in the reduction process demands the strong adsorption of CO_2 on the catalytic surface, which is subsequently followed by its activation by the surface active sites.[16-18] The CO_2 photoreduction imitates the photosynthesis of plants where it utilizes water and CO_2 and convert them into chemical fuels such as methane, methanol and value added multi carbon products. The reaction product ratio will vary with number of electrons participating in the reaction, nature of photocatalytic materials and the adopted reaction conditions.[14-18]

The primary step in the semiconductor photocatalysis involves the absorption of light energy by the catalysts greater than or equal to its band gap to generate electron-hole pairs. These excitons migrate towards the surface and interacts with the adsorbed species to produce the products *via* redox reactions, which later desorb from the surface of the catalyst.[19] As per
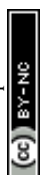
1 theoretical calculations and kinetic studies, it is reported that the CO, COOH and COH are

2 common intermediates for CO_2 reduction reaction.[20] The CO_2 photoreduction will proceed only

3 when the energy of the conduction band of the concerned semiconductor is more negative than

4 the reduction potential of the desired products.[2] Semiconductors such as BiOX, CdS, g-C_3N_4,

5 Cu_2O, SnO_2, TiO_2, ZnO, and perovskites are extensively used for CO_2 reduction reaction.[21-31]

6 The bismuth-based materials has garnered major interests as they possess band gap spanning

7 the major portion of the solar spectrum and band edge positions are convenient to trigger

8 desired reduction and oxidation reactions.[32-37] The Bi 6s orbital hybridize with O 2s orbital to

9 form highly dispersive energy band structure which promotes charge carrier transport and also

10 exhibits high oxidizing capabilities. Furthermore, bismuth based materials offer stable

11 structure with layered spaces to promote the intercalation of foreign ions without changing

12 their structural features. Being less toxic, environmentally friendly and remarkably

13 photostable, the distinct morphology of bismuth-based materials have been explored for

14 photocatalytic applications.[11, 38] Different categories of bismuth based materials such as unitary

15 (Bi), binary (Bi_2S_3, Bi_2O_3), and ternary (stoichiometric and non-stochiometric bismuth oxy

16 halides, $BiVO_4$, Bi_2WO_6, Bi_2MoO_6) are used for photocatalytic CO_2 reduction.[36, 39-41] However,

17 massive charge carrier recombination and narrow optical response of single phase

18 photocatalysts limits their performance under the light illuminated conditions.[39, 42, 43] Several

19 reviews on bismuth-based materials towards photocatalytic applications can be found in the

20 literature.[37, 38, 41, 44, 45] However, recent trend in the interfacial engineering of these materials is

21 scarcely emphasized. Motivated by these aspects, this review summarizes the interfacial

22 engineering of bismuth-based materials with various semiconductors under different reaction

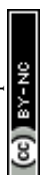23 conditions and their performance towards photocatalytic CO_2 reduction are outlined.

24

25
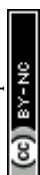
2.0 Strategies for interfacial engineering

Heterojunction of semiconductors found to be a promising method to improve the photocatalytic performance owing to the synergistic interactions arising from the structure-optical-morphological properties of the integrated semiconductors.[46-49] The prime requisite to fabricate a heterojunction is that the band position of the concerned semiconductors must possess different potentials to achieve maximum separation of the charge carriers.[50, 51] Secondly, the bandgap response of the individual components must be substantially different so that the composite absorbs larger fraction of the solar light for the photocatalytic reactions. The various types of heterojunctions have been fabricated for bismuth-based materials to increase the conversion efficiency of CO_2 to value added products. Following are the methods that have been adopted extensively for various types of bismuth-based materials:

(i) Type-II heterojunction: When two semiconductors with different band edge potentials and Fermi levels encounters each other, three types of heterojunctions such as Type-I, Type-II and Type-III are formed. The Type-I allows for the migration of both the charge carriers from one semiconductor to another owing to their straddle bandgap configuration, which enhances the charge carrier recombination (Figure 1). In contrast, Type-II is efficient in separating the electron-hole pairs due to the staggered configuration of the electronic band positions. It is also referred to as cascade electron transfer process, as it promotes interfacial charge carrier separation. However, the potential of electrons and holes decreases lowering the efficiency of the composite.

(ii) Z-scheme: Though type II heterojunction can improve the photocatalytic activity, they suffer from recombination of electron and holes. This disadvantage can be avoided by Z-scheme technique.[52]
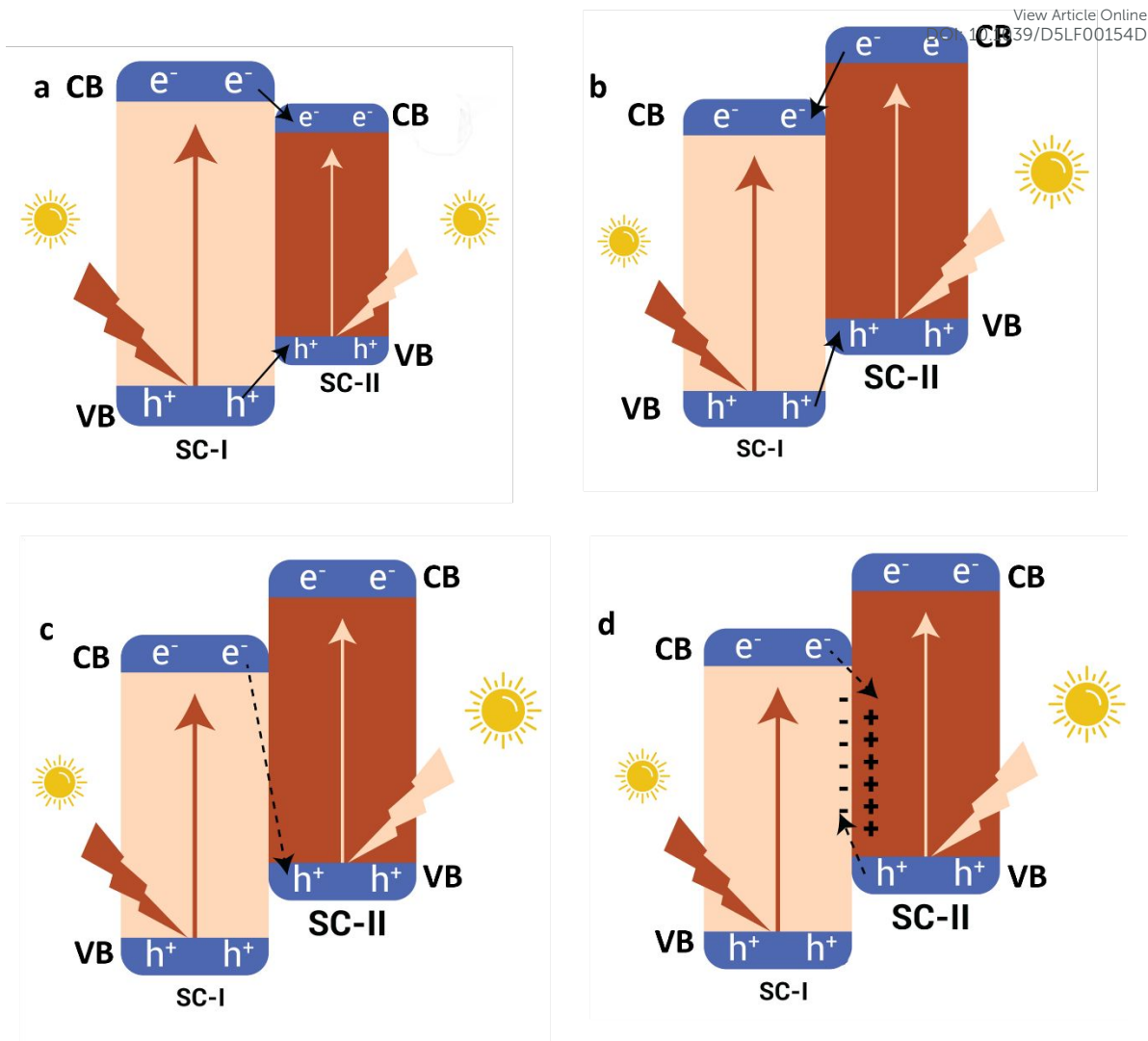
1

Figure 1. Different types of semiconductor-semiconductor heterojunction: a) Type-I, b) Type-

II, c) Z-scheme, and d) S-scheme heterojunction. Note: SC = semiconductors; CB = conduction

band; VB = valence band.

Band gap configuration in Z-scheme is similar to Type-II configuration, but an exception

remains in the direction of electron transfer pathways. The less potential electrons accumulated

on one side of the semiconductor recombine with less potential holes of another

semiconductor, thereby retaining the high potential electrons and holes for the redox reactions.
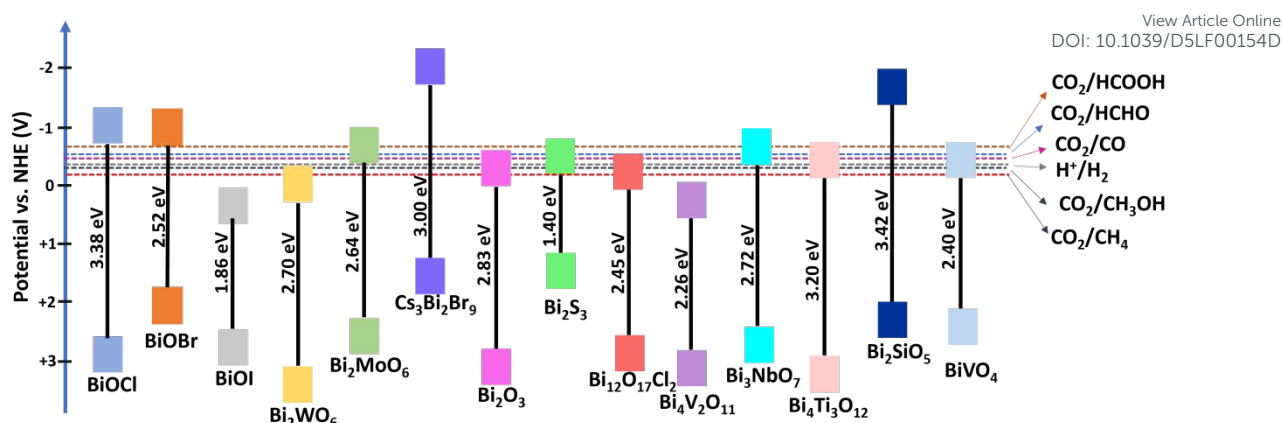
9

5

Figure 2. Band positions of different bismuth based semiconductors.

Though, significant improvement can be observed from Z-scheme photocatalysis, this technique also fails to deliver superior photocatalytic performance as the limitation arises from the choice of the semiconductors.[53]

(iii) S-Scheme heterojunction: Step scheme (S-scheme) was introduced by Jiaguo Yu group.[54, 55] When reduction and oxidation photocatalysts are in close contact, the electrons in the oxidation photocatalyst recombines with holes present in the valence band of the reduction photocatalyst. The positive and negative charge is formed at reduction and oxidation photocatalyst and the direction of in-built electric field emerges from reduction to oxidation photocatalyst. Thus, the electron depletion at the reduction photocatalyst causes the upward band bending and electron accumulation at the oxidation photocatalyst results in the downward band bending. Driven by the rise of electric field and the Coulombic force of attraction, the electrons in oxidation photocatalyst recombine with holes in the reduction photocatalyst, while preserving the energetic charge carriers for the desired redox reactions.[55] The preserved electrons and holes are available for reaction to take place making the photocatalytic system to be more efficient than Type-II and Z-scheme heterojunction. The position of conduction and valence band for different bismuth based materials clearly signifies their ability to participate in the CO_2 reduction reactions (Figure 2).

1　　　(iv) Cocatalyst loading

2　　　The cocatalysts such as noble metals which include Pt, Ag and Au, non-noble metals

3　such as Bi, Cu, and Mg have been exploited to improve the performance of the photocatalytic

4　process.[56-58] The activation energy required to convert CO_2 to any value-added products can be

5　reduced with the addition of cocatalysts on the semiconductor surface. It is also found from the

6　literature that the it increases the reaction site and play a vital role in determining the selectivity,

7　adsorption of CO_2, and photostability of the photocatalyst.[59, 60]

8　　　(v) Defect engineering: Defects are of paramount importance in tailoring the bandgap

9　and charge carrier dynamics of functional semiconductors. In general, anionic vacancies like

10　oxygen vacancies can narrow the gap region and shift the optical response of the host matrix

11　to the visible region. Few reports also hints that the tailoring the bandgap depending on the

12　density of oxygen vacancies in the host matrix.[61, 62] Furthermore, oxygen vacancies can also

13　trap the charge carriers to restrain their recombination with holes and extends the charge carrier

14　lifetime. On the other hand, cationic vacancies can temporarily trap the holes. In recent times,

15　vacancy engineering for Bi-based materials is reported to show superior photocatalytic activity

16　compared to less defective counterparts.[45, 63, 64] These defects can alter the lattice strain and

17　surface defects which can facilitate the adsorption of oxygen, to promote the formation of

18　superoxide radicals in the solution phase.[65]

19

20　**3.0 Bi-based semiconductors**

21　**3.1 BiOX**

22　　　BiOX is highly crystalline with tetragonal crystal structure and extensively used for

23　photocatalytic applications due to its layered structure and narrow band gap.[66] It is composed

24　of layered structure with $[Bi_2O_2]$ slabs and two arrays of $[X]^-$ ions. The exciton pairs can be

25　effectively separated as the charge density is higher in Bi-O than that of [X] slabs. In addition,
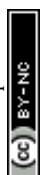
1 static electric field along the *c*-axis avoids the massive charge carrier recombination View Article Online DOI: 10.1039/D5LF00154D

2 pathways.[67] The BiOCl is an indirect band gap semiconductor which decreases the chance of

3 recombination of the charge carriers.[68-71] Defects in BiOX can influence the photocatalytic

4 activity by modifying charge transport and electronic structure of the material. The oxygen

5 vacancy can be introduced easily to the BiOCl as Bi-O bond is weak. These oxygen vacancy

6 can increase the number of active sites to facilitate the CO_2 adsorption and activation process.[72]

7 It has been observed that the majority of BiOX based heterojunctions produce only CO and

8 CH_4 and therefore, interfacial engineering of BiOX plays a crucial role in improving the

9 selectivity.[73] The crystal facet of BiOCl also influence the photocatalytic activity when coupled

10 with CdS. The CdS/BiOCl with exposed facet exhibited higher performance compared to

11 facets.[67] This was attributed to the effective coupling of interfacial electric field of the

12 heterojunction and the polarization electric field of {001} crystal facet of BiOCl. The CH_4 and

13 CO production with {001} -BiOCl/CdS was higher compared to the (zero one zero)-

14 BiOCl/CdS, despite the bandgap energy and band edge potential of both the facets were almost

15 similar. This result suggests that the crystal facet orientation of BiOCl has profound influence

16 on the adsorption and activation of CO_2. As the polarization electric field for BiOCl is along

17 the *c*-axis, excited electrons easily accumulate on the {001} facet, which favour their

18 recombination with valence band holes of CdS, thereby improving the energetic electrons and

19 holes from CdS and BiOCl respectively to participate in the redox reactions.[67] The 2D/2D Vo-

20 WO_x/BiOCl S-scheme has been fabricated by a self-assembly process. S-scheme

21 heterojunction not only offered separation of photogenerated charge carriers but also provided

22 dual interfacial pathway to improve the charge separation process.[74] Jiang *et al*. showed that

23 the CO_2 molecules adsorbed on BiOCl (001) and WOx (010) surface.[75] The study also

24 suggested that CO_2 molecules are more stable on BiOCl surface than that of WO_x. The

25 conduction band level for WO_x is not sufficient to carry out reduction of CO_2 molecules. As a

1   result, the adsorption and reaction sites are available only on the surface of BiOCl.[75] Similarly,

2   Bi_2WO_6 has also been used to fabricate the heterojunction with BiOCl. Unlike S-scheme as in

3   WO_x/BiOCl, they formed typical Type-II heterojunction. Theoretical calculations confirm that

4   the production of CHO* intermediate on the surface of BiOCl/Bi_2WO_6 is responsible for the

5   selective generation of CH_4. The high selectivity towards CH_4 is due to ultrathin nanosheet

6   morphology (2D/2D) and the presence of internal electric field at the heterointerface.[76]

7   BiOCl/In_2O_3 fabricated by hydrothermal method with Type-II heterojunction yielded with high

8   rate of CO_2 photoreduction.[77] Enhanced performance was attributed to the narrow band gap

9   which falls in the visible region. In another report, Chlorophyl and Mg co-modified BiOCl

10 (Chl-Mg/BiOCl) microsphere exhibits superior photocatalytic applications. Chl-Mg/BiOCl

11 yielded 100 μmol g^{-1} of CH_4. Chlorophyl on the surface of the BiOCl produces singlet states of

12 Chl-Mg* which acts as an electron donor. Mg forms a complex with chlorophyl and helps in

13 improving the stability of the catalyst.[78]

14   It is also noted that cocatalyst loading on the photocatalyst is a potential approach to

15 improve the adsorption of the intermediates in CO_2 reduction there by influencing the coupling

16 of C-C bond. To improve C_2 hydrocarbon selectivity and to enhance photocatalytic reduction

17 of CO_2 transition metal cocatalyst with nanostructures have been loaded on BiOCl. [71, 79, 80]

18 Selectivity towards CH_4 was increased by decorating with noble metal such as Pd on BiOCl.

19 Pd has not only decreased the recombination of the charge carriers but also changed the

20 intermediate formed (from HCO_3^- to HCOO* and CH_3O^*, CH_3O^* will be reduced to CH_4) on
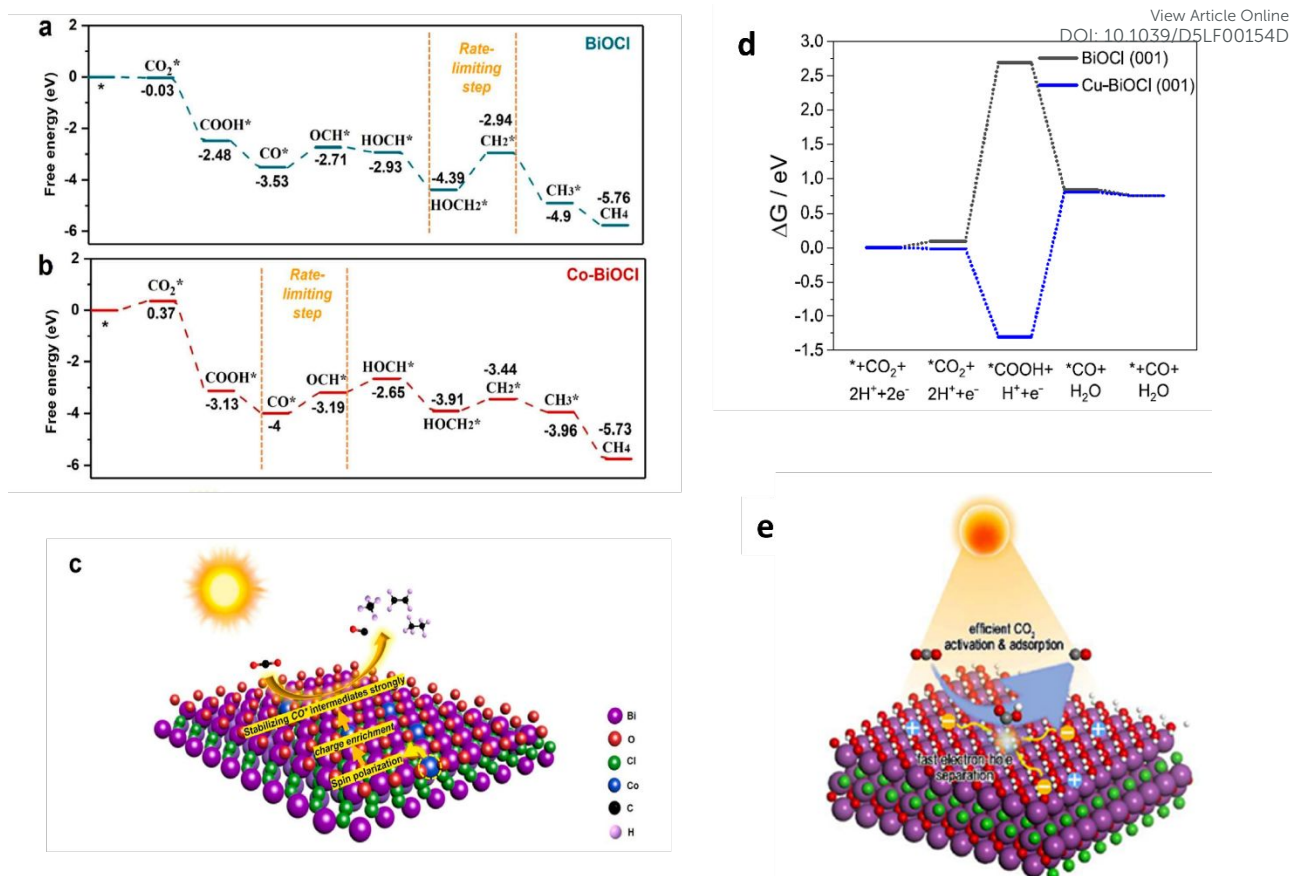
21 the surface of BiOCl.[80]
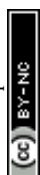
Figure 3. Computed Gibb's free energy main reactions for photocatalytic reduction of CO_2 on a) pristine BiOCl b) Co-BiOCl [79] d) Gibb's free energy diagram for CO_2 reduction in Cu-BiOCl e) scheme for photoreduction of CO_2 to CO. Reproduced with permission from ref. 68. Copyright 2022, American Chemical Society and ref. 79. Copyright 2024, Elsevier.

In situ diffused reflectance infrared Fourier transformation (DRIFT) revealed that the key intermediates are bidentate formate and methoxy which promotes the selectivity for CH_4 production.[80] Photocatalytic activity of BiOCl ultrathin sheets can be altered by introducing magnetically active metal particles. In this context, Li *et al.* reported that Co doped BiOCl selectively converts CO_2 to CH_4 and C_{2+} products which was not possible with the pristine BiOCl. The increase in the photocatalytic activity was due to Co-O bond on the Co-BiOCl.[79] The increased CO_2 adsorption is primarily responsible for the enhanced CO_2 photoreduction. It is also found that the introduction of Co on BiOCl generates spin polarized charge and

1 changes the intermediate formed on the surface of BiOCl. In order to generate CH_4 and C_{2+}

2 products, CO^* intermediate will be formed via the formation of $COOH^*$ intermediate (Figure

3 3a-c). The rate limiting step for Co-BiOCl is the conversion of CO^* to CHO^* whereas for

4 BiOCl $HOCH_2^*$ to CH_2^* was the rate limiting step.[79] This work shows that the spin polarization

5 of the charge on the surface of BiOCl has resulted in the formation of stable intermediate and

6 promoted C-C coupling to yield CH_4 and C_{2+} products. Similarly, Cu was modified on BiOCl

7 nanosheets which resulted in surface charge distribution. Cu-BiOCl has improved the catalytic

8 performance and enhanced the absorption of light.[71] Also, it has decreased the activation energy

9 barrier, which facilitates the faster consumption of CO_2 molecules resulting in increased CO

10 yield (Figure 3d-e). Metal cotalysts such as Pt, Pd and Au metals have been decorated on BiOCl

11 nanostructures by deposition-precipitation method. The main product obtained with BiOCl is

12 CO, while CH_4 production dominated after loading the metal cocatalyst.[81]

13 Chen *et al.* demonstrated that Z-scheme In_2S_3/BiOCl can be prepared by solution combustion

14 synthesis. The nanopetals of BiOCl were decorated on the nanoflowers of In_2S_3 in solution

15 combustion synthesis. In_2S_3/BiOCl nanostructures are evenly deposited on the surface of the

16 nanowires of MnO_2.[82] Initially, In_2S_3 nanoflowers was synthesized by hydrothermal approach.

17 The synthesized In_2S_3 nanoflowers were dissolved in thiourea and the bismuth nitrate precursor

18 along with concentrated nitric acid stirred vigorously. To this solution urea and ammonium

19 chloride was added and stirred constantly for 24 h. The In_2S_3/MnO_2/BiOCl was fabricated by

20 using the suspension of In_2S_3/BiOCl by hydrothermal approach. Despite having higher

21 solubility than In_2S_3, BiOCl grows on the surface of the In_2S_3.[82] In_2S_3/BiOCl deposits uniformly

22 on MnO_2 nanowires which forms an assembly of hierarchical structures. It is noted from this

23 work that the S^{2-} ions play a significant role in controlling the nucleation of In_2S_3, forming the

24 morphology and controlling the size of the nanostructure. This ternary heterojunction exhibited

25 photocatalytic reduction rate of CO_2 to CO (51.2 $\mu mol\ g^{-1}h^{-1}$), C_2H_4 (63.2 $\mu mol\ g^{-1}h^{-1}$), and CH_4
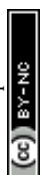
1   (42.4 μmol g^{-1}h^{-1}).[82] The observed synergistic effect is attributed to the Mn $3d$ electrons which

2   has narrowed the band gap. The key intermediates of the photocatalytic reduction were CHO*

3   and COOH* for the production of CO, CH$_4$ and C$_2$H$_4$.[83] BiOCl with BiOBr heterojunction was

4   fabricated by mechanical method which yielded Type-I heterojunction.[84] This heterojunction

5   reduced CO$_2$ to CO with the formation rate of 7.353 μmolg^{-1}h^{-1}.

6   CsPbBr$_3$ perovskite is attracting increased attention due to its visible light harvesting

7   capability and its high conduction band position enhances the reduction capability.[85] Various

8   heterojunctions have been constructed based on CsPbBr$_3$ perovskite as the pristine material

9   suffers from severe recombination of charge carriers. A 2D/2D S-scheme heterojunction

10 comprising BiOCl and CsPbBr$_3$ was fabricated by an efficient self-assembly method. The zeta

11 potential measurements revealed the negative and positive surface charge density on BiOCl

12 and CsPbBr$_3$ which can adhere to each other through electrostatic force of attraction. The

13 nanosheets of perovskite were assembled on the nanosheets of BiOCl which provides face-to-

14 face contact and strong interfacial area existed between them. This heterojunction gave much

15 improved photocatalytic activity towards CO$_2$ reduction to form CO as major product and CH$_4$

16 as minor product.[85] The high content of CsPbBr$_3$ in the composite reduced the CO production,

17 while the yield of CH$_4$ remained almost same irrespective of its content.[85]

18 Chen et al. synthesized BiOCl by hydrothermal method and the oxygen vacancy was

19 introduced by heating at 60 ℃ in the vacuum oven.[86] The Z- scheme heterostructure was

20 fabricated by vigorously stirring powders of BiOCl with oxygen vacancy and 2D g-C$_3$N$_4$

21 nanosheets in ultrasonic bath. It is found that the heterojunction can be constructed effectively,

22 if the two individual components are stirred in the ultrasonic bath rather than on the magnetic

23 stirrer.[86] Authors have observed that photocatalytic reduction of CO$_2$ to CO for interfacial

24 oxygen vacancy induced g-C$_3$N$_4$/BiOCl heterostructure was 1.6 times more than the pristine

25 g-C$_3$N$_4$/BiOCl.
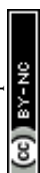
1 Ternary composites of BiOCl with dual Z- scheme of $BiOCl/g-C_3N_4/Ag_2CrO_4$ has also been

2 synthesized and found to give CO and CH_4 with yields of 30.20 and 81.21 $\mu mol\ g^{-1}$

3 respectively.[87] It is possible to make ternary heterojunction with C_3N_4 and BiOCl using

4 Ag_2CrO_4 photocatalyst. The dual Z-scheme heterojunction photocatalyst was synthesized by

5 simple method. $BiOCl/C_3N_4$ was prepared by taking the Bi, C and N precursors and heating

6 solvothermally using ethylene glycol as solvent for 12 h at 140 °C. However, The $BiOCl/C_3N_4$

7 $/Ag_2CrO_4$ was synthesized by stirring the powders of $BiOCl/C_3N_4$ and precursors of Ag^+ and

8 Cr^{6+} ions for 2 h. When the ternary heterojunction is formed, the morphology of BiOCl, C_3N_4

9 and Ag_2CrO_4 was retained. The irregular shapes of Ag_2CrO_4 were uniformly decorated on the

10 microspheres of BiOCl and nanosheets of C_3N_4. It is observed that the C_3N_4 provides charge

11 transfer pathway to form dual Z-scheme heterojunctions.

12 $MgIn_2S_4$ is another chalcogenide, which exhibits promising band gap and photostability which

13 is extensively used for removal of liquid pollutant and for the production of hydrogen.[88] Zhang

14 *et al.* fabricated the S-scheme heterojunction of $Bi-BiOCl/MgIn_2S_4$ by hydrothermal technique

15 and $NaBH_4$ reduction technique.[89] The microspheres of BiOCl with average diameter of 1 μm

16 was obtained from hydrothermal method and Bi self-doped BiOCl was synthesized by treating

17 BiOCl micorspheres with $NaBH_4$ followed by vacuum drying. $Bi-BiOCl/MgIn_2S_4$ was

18 synthesized by *in situ* hydrothermal method and the obtained product exhibited ultrathin sheets

19 of BiOCl dispersed on the surface of $MgIn_2S_4$ marigold flower. It is noted that $NaBH_4$ which

20 is used as a reducing agent has influenced the morphology of BiOCl and caused the rough

21 surfaces and loose internal structures.[89] The formed S-scheme heterojunction exhibited the

22 conversion of CO_2 to CH_4 with the yield of 25.72 $\mu mol/g$.
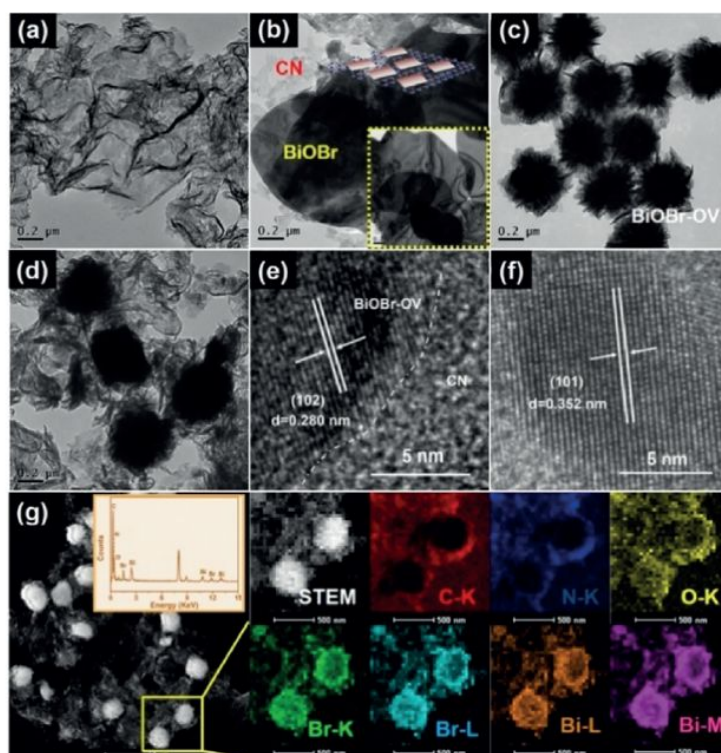
23

Figure 4. Transmission electron microscope (TEM) images of a) pure C_3N_4 (CN), b) CN-BiOBr, and BiOBr marked in the lower right corner, c) bare BiOBr-OV, d) CN-BiOBr-OV catalyst, e), f) High resolution (HR)-TEM images of CN-BiOBr-O_V heterojunction with the lattice spacings, and g) elemental maps of C, N, O, Br, and Bi in the CN BiOBr- O_V photocatalyst. Reproduced with permission from ref.[91]. Copyright 2020, Wiley-VCH.

BiOBr exhibits appropriate band structure, making it suitable for photocatalytic applications. However, the performance of BiOBr needs modification in the band structure as it promotes electron-hole recombination. The g-C_3N_4 has band gap that falls in visible light wavelength and therefore coupling with BiOBr would benefit to use larger fraction of solar light.[90] The hollow microsphere morphology of BiOBr was completely destroyed when grown over the surface of g-C_3N_4 sheets under wet-chemical approach. The composite was selective to produce CH_3OH from CO_2 at the optimized content of g-C_3N_4.[90] Liu *et al.* synthesized BiOBr/C_3N_4 by ultrasonicating the synthesized BiOBr and C_3N_4 for 6 h.[91] BiOBr was

1　synthesised by solvothermal technique and the oxygen vacancy was introduced by adding the

2　surfactant along with precursors. Furthermore, oxygen vacancy can be introduced to BiOBr by

3　adding the surfactant in solvothermal synthesis which is not possible in hydrothermal synthesis.

4　The g-C_3N_4 heterojunction was constructed with BiOBr possessing surface oxygen vacancy.

5　The morphology has been compared in figure 4.[91] Along with heterojunction interface, the

6　presence of surface oxygen vacancy has promoted the charge separation and hence

7　photocatalytic performance was enhanced.

8　Xi *et al.* also synthesized BiOBr/Bi_2S_3 and conducted a detailed analysis on the growth

9　mechanism of Bi_2S_3 nanoarrays on BiOBr nanoplates. [92] Facet sensitive growth of Bi_2S_3 can be

10　used as an effective strategy to construct a heterojunction with increased number of active sites

11　using thioacetamide as the sulfur source. It can also be observed that the structurally a very

12　close matching of the lattice parameters *c*-axis of Bi_2S_3 and *a* or *b* axis of BiOBr facilitates the

13　facet selective growth of Bi_2S_3.[92] The established S-scheme heterointerface between BiOBr and

14　Bi_2S_3 and produced 20.32 μmol g^{-1} CO from CO_2 photoreduction.[92] The large area exposure of

15　both semiconductors helps in exploiting maximum amount of visible light. Directional growth

16　of Bi_2S_3 as mesh like nanostructure on the surface of the BiOBr nanoplates *via* topotactic

17　transformation process.[92]

18　Vertically aligned Bi_2S_3 nano walls help in minimal shielding of the light falling on the surface

19　of BiOBr nanoplate substate which extends the optical response to 1000 nm (Figure 5). Authors

20　have demonstrated that vacancy rich BiOBr substrate enhances the adsorption of CO_2 on the

21　surface.[92] Due to weak Bi-O linkage, it is easy to produce oxygen vacancy in BiOBr. Since

22　oxygen vacancy creates trap centres for photogenerated charge carriers, it enhances the

23　separation efficiency. Rational design of oxygen vacancy rich BiOBr heterojunction would be

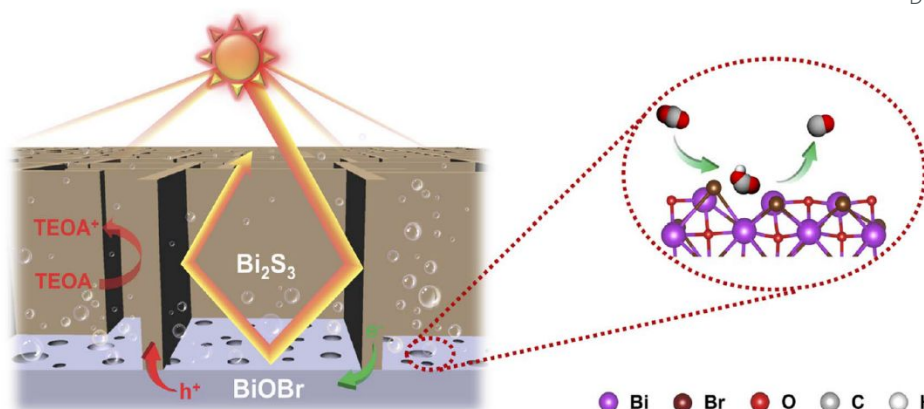24　beneficial in elevating the photocatalytic CO_2 reduction.

25
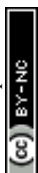
Figure 5. Schematic illustrating the proposed mechanism for the photocatalytic CO_2 reduction over the BiOBr/Bi_2S_3 heterojunction. Reproduced with permission from ref. 92. Copyright 2022, Royal Society of Chemistry.

Ma *et al.* suggested that instead of constructing the conventional heterostructures, co-sharing atoms can be effective strategy for interfacial engineering along with suitable band structure. This work stresses on the atom level interfacial contact of the heterostructures.[93] BiOBr is coupled with Bi_2S_3 a layered structure which delivers full spectrum responsive S-scheme photocatalyst. BiOBr/Bi_2S_3 was fabricated by *in situ* ion exchange method, which yielded Bi_2S_3 nanorods on BiOBr nanosheets. It is observed that the morphology of BiOBr nanosheets changed after reacting with sulphur atoms in hydrothermal reaction which indicates the occurrence of ion exchange reaction led to the formation of the heterojunction. It is also noted that as the sulphur concentration increases the density of Bi_2S_3 nanorods on BiOBr nanosheets. BiOBr/Bi_2S_3 heterostructure yielded CO with 20.32 μmol g^{-1}.

Bi_2WO_6 a layered structure, comprising of fluorite like $[Bi_2O_2]^{2+}$ and perovskite like $[WO_4]^{2-}$. Owing to its strong oxidising capability of valence band, Bi_2WO_6 is extensively studied for photocatalytic applications. However, the performance is limited by severe recombination of the photogenerated charge carriers. Therefore, composite of Bi_2WO_6 and BiOBr would

1 enhance the photocatalytic conversion of CO_2 reduction.[94] It is known from the literature that

2 the surface oxygen vacancy can easily be created on Bi_2WO_6 and BiOBr which can provide

3 increased number of adsorption sites for the reactants and trap states for the photogenerated

4 charge carriers.[68, 95] Apart from oxides and sulphides, layered double hydroxides (LDH) were

5 also used for making the composites with BiOBr. The CoAl-LDH has high selectivity towards

6 CH_4 whereas NiAl-LDH yields CO.[96] As CoAl-LDH has suitable band positions with BiOBr,

7 facile electron hole pair separation is possible. The heterojunction was fabricated by

8 hydrothermal method. The optimal amount of ultrathin CoAl-LDH loaded on BiOBr has

9 improved the selectivity towards CH_4, a value-added chemical.[96] In most of the cases, pristine

10 BiOBr yields CO from photocatalytic conversion. In order to obtain value added products such

11 as CH_4, the *p*-type semiconductor Cu_2O was heterojunctioned with BiOBr.[97] Recent studies

12 have found that using metallic Bi found to be beneficial in enhancing the photogenerated

13 charge separation.[14, 98, 99] Many reports claim that Bi-BiOBr show enhanced performance

14 towards CO_2 photoreduction.[100] Rich oxygen vacancy can be successfully synthesized by *in*

15 *situ* partial ion-exchange method on the heterojunction materials.[74] More importantly, metallic

16 Bi self-doped Bi_2SiO_5 synthesized by hydrothermal method can provide facile situation for S-

17 scheme heterojunction and thereby increase the redox capability in S-scheme BiOBr-

18 (001)/Bi_2SiO_5/Bi heterojunction which gives CO production rate of 234.05 μmmol $g^{-1}h^{-1}$.[74]

19   AgBr is another excellent photo-response material, but it suffers from instability due to

20 reduction of silver ion to metallic silver upon exposure to the light. The low electron-hole

21 recombination and extended visible light response was achieved in BiOBr/AgBr.[101] Density

22 functional theory and experimental results have demonstrated that electron depletion region is

23 produced on BiOBr whereas electrons accumulate on AgBr resulting in S-scheme

24 heterojunction to produce CO and CH_4. As intensity of the light increases the production of

25 CO and CH_4 increases linearly proving that the reaction follows first order kinetics.[101]

1     $BiOBr_xI_{1-x}$ was coupled with $BiPO_4$ to form *p-n* junction which was synthesized by

2   solvothermal method.[102] This heterojunction exhibited increased photocatalytic activity

3   compared to $BiPO_4$.[102] The 0D/2D $CsPbBr_3$/BiOBr grown under different conditions were

4   selective to produce CO rather than CH_4.[103, 104] The oxygen vacancies in BiOBr modulated the

5   adsorption and activation of CO_2 and highly negative conduction band position of QDs

6   promoted the reduction reaction pathways. Recently, coupling of BiOBr with *p*-type NiO was

7   reported to be active for CO_2 reduction to yield CO and CH_4.[105] It was proposed that adsorption

8   of CO_2 on the composite surface forms carbonate species, which is converted to formic acid by

9   capturing protons and electrons. The formic acid was successively transformed to

10   formaldehyde, methanol and methane by continuous hydrogenation process. Alternatively,

11   CO_2 was directly transformed into CO by hydrogenation and dehydration.[105] The conduction

12   band of NiO is mainly derived from the orbitals of Ni which enables the accumulation of

13   electrons at the Ni sites of NiO, which primarily serves as adsorption centres for CO_2.[105]

14   Among BiOX materials, BiOI exhibits narrowest band gap (~1.6 to 1.9 eV) which enhances

15   the absorption of light in visible region. BiOI consists of alternative layers of $[Bi_2O_2]$ and iodine

16   held by Van der Waal's force of attraction. BiOI, an indirect band gap semiconductor was

17   heterojunctioned with a perovskite material to improve the interfacial charge separation. A

18   direct band gap semiconductor caesium formamidinium lead halide perovskite

19   ($Cs_{1-x}FA_xPbBr_3$) was coupled with nanosheets of BiOI nanomaterial.[106] It is worth noting that

20   the heterojunction was created by stirring the solution of BiOI and perovskite nanocrystals due

21   to the low surface energy of perovskite nanocrystals and strong interaction between Bi cation

22   and perovskites's Br anion. Perovskite with cuboid structure is well decorated on the ultrathin

23   nanosheets of BiOI as confirmed by SEM and TEM images. Authors have demonstrated the

24   kinetic models in figure 6 and they claim that BiOI have more surface defects than that of

25   perovskite $Cs_{1-x}FA_xPbBr_3$ (CF).[106] However, when BiOI and CF were heterojunctioned, the

1   blue side of the photoinduced absorption band (PIA) becomes allowed transition with Bi^{+3}

2   doping in CF. Internal electric field (IEF) is created due to the electron transfer between BiOI

3   and CF which facilitates the recombination of electrons from the conduction band of the BiOI

4   and holes of the valence band from CF perovskite to form S-scheme heterojunction as depicted

5   in figure 6.

6   Formation of S-scheme was confirmed by X-ray photoelectron spectroscopy. Femtosecond

7   transient absorption spectroscopy was utilized to investigate the ultrarapid dynamics of charge

8   carriers. Doping of Bi^{+3} has created deep defect states at the interface which degraded the

9   photocatalytic activity of CO_2 reduction.
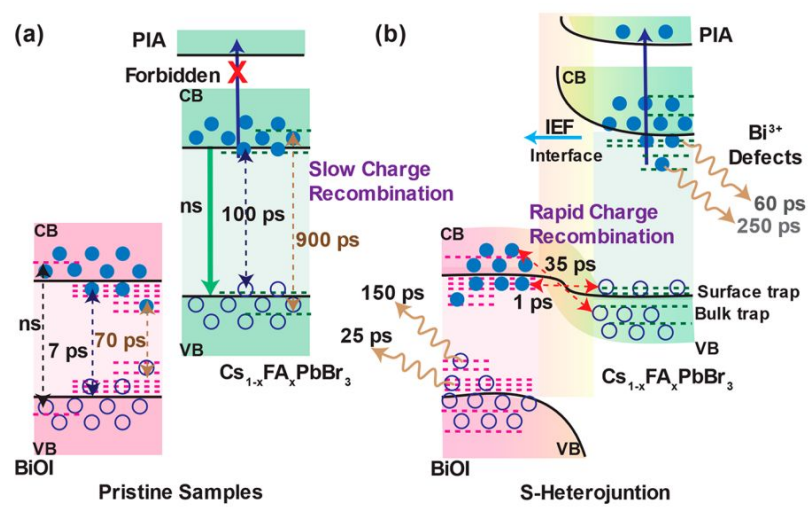
20   **Figure 6.** (a) Depiction of kinetic models for (a) pristine BiOI and $Cs_{1-x}FA_xPbBr_3$ (CF) and (b)

21   hybrid $Cs_{1-x}FA_xPbBr_3$:BiOI (CF:BiOI) heterojunction samples. IEF represents internal electric

22   field generated in the heterojunction. Reprinted with permission from ref.106. Copyright 2022,
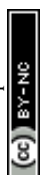
23   American Chemical Society.

1   In another work, BiOI nanosheets and In_2O_3 were synthesized individually by solvothermal

2   technique.[107] Similar to the above work, heterojunction was formed by mixing the as

3   synthesized semiconductors and heating solvothermally to obtain the heterojunction of

4   BiOI/In_2O_3. The spherical morphology of In_2O_3 was decorated on the nanosheets. However, it

5   is observed that the morphology was non-uniform throughout the heterojunction material. It

6   can be inferred that heterojunction fabricated from solvothermal technique from taking the

7   individual semiconductor materials yielded aggregation of In_2O_3 on BiOI nanosheets.[107] The

8   BiOI/C_3N_4 constructed on hydrophobic carbon fiber paper through electrophoretic deposition

9   process exhibited significant selectivity for CO production from CO_2 reduction reaction.[108]

10 Initially, pre-formed g-C_3N_4 sheets and iodine powder were dispersed in acetone solvent,

11 wherein protons were released upon reaction of iodine with acetone. Later, BiOI/carbon fiber

12 and Pt electrodes was used as cathode and anode respectively, which was followed by

13 application of external potential.[108] The BiOI-nanosheets were vertically grown on the surface

14 of carbon fibers and g-C_3N_4 sheets completely wrapped the fibers to form 2D/2D contacts

15 between the semiconducting surfaces.[108]

16 Authors have carried out interfacial engineering with the formation of *p-n* junction by

17 compositing BiOI and Zn_2TiO_4.[109] The presence of oxygen vacancies speeds up the CO_2

18 reduction process.[110] It extends the lifetime of the charge carrier and also acts as activation sites

19 to enhance the adsorption of CO_2. Wang *et al.* have fabricated 2D/2D Bi_2MoO_6/BiOI S-scheme

20 heterojunction to enhance the photoreduction of CO_2.[110] BiOI nanosheets were grown on few

21 layered Bi_2MoO_6 nanosheets using solvothermal technique. S-scheme mechanism was

22 investigated by utilizing time resolved photoluminescence spectra, work function and charge

23 density difference analysis. Nanosheets of Bi_2MoO_6 on BiOI nanospheres demonstrated that

24 the Van der Waal's heterojunction is effective in accelerating the photogenerated charge
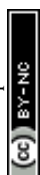
1 carriers. Authors have established that the charge transfer takes place from Bi_2MoO_6 to BiOI

2 in addition to large area Van der Waal's heterojunction and S-scheme heterojunction.[110]

3 Hongyu Fu *et al.* have synthesized the heterostructure of Cu_2O and carbon loaded BiOI using

4 two steps methods.[111] The yield of methanol and ethanol from CO_2 was 722.8 μmol g^{-1} for 8 h.

5 Photocatalytic performance of C-BiOI was significantly increased compared to pristine BiOI.

6 Heterostructure of Cu_2O and carbon loaded BiOI further improved the photocatalytic activity.

7 Cu_2O is a *p*-type semiconductor with the band gap of 2.2 eV. Nanoparticles of Cu_2O were

8 decorated on to the nanosheets of C-BiOI by chemical deposition method. The introduction of

9 Cu_2O remarkably increased the photogenerated charge carriers and also the surface area of the

10 heterostructure which provided more active sites for CO_2 reduction.[111]

11 BiOI based 2D-2D heterojunction materials have been designed in order to elevate the

12 photocatalytic performance. Band structure of C_3N_4 and BiOI is favourable to form S-

13 heterojunction, which can boost the charge separation.[71] Li *et al.* have proposed S-scheme

14 charge transport which increases the redox capability of both BiOI and C_3N_4 and converts CO_2

15 to CO.[112] Formation of S-scheme heterojunction facilitates the electrons to move from C_3N_4 to

16 BiOI creating depletion region in C_3N_4 whereas electron accumulation occurs at BiOI. In order

17 to develop wide wavelength responding photocatalysts, authors have interfaced In_2O_3 with

18 BiOI as In_2O_3 exhibits band gap of 2.8 eV which falls in the visible region. It is also known to

19 possess good electronic conductivity and resistant to corrosion. Type-II heterojunction was

20 fabricated by solvothermal method.[107] In_2O_3/BiOI produces CO and CH_4.[72] 2D heterojunction

21 of an atomic thin BiOI/BiOCl was constructed to mitigate the charge recombination. Oxygen

22 – iodine binary vacancy was created which decreased the oxidation energy barrier and formed

23 *COOH, a stable rate limiting intermediate.

24

25

3.2 BiVO$_4$

As there is a larger difference in the reduction potential for CO_2 and the conduction band potential of BiVO$_4$, only few studies are available on pure BiVO$_4$ towards the photocatalytic reduction of CO_2. It is found that BiVO$_4$ can selectively convert CO_2 to ethanol. In order to improve the photocatalytic performance, several strategies is adopted such as morphology control, doping with metals and defect introduction.[56, 113] It is found from the literature that metal nanoparticle doping can effectively tune the band gap of BiVO$_4$ and there by improve the selectivity and conversion efficiency.[113, 114] The high performing BiVO$_4$ quantum dots with Au nanoparticles were fabricated and heterojunctioned with rutile nanorod arrays.[115] The surface plasmon resonance of Au has not only helped in boosting the charge separation but also increased the light response of the photocatalyst.[116, 117] This led to the prolonged life time of the charge carriers and the increased separation efficiency. Similar approach was adopted to construct Z-scheme BiVO$_4$-Au-Cu$_2$O, where BiVO$_4$ (010)-Au provides Schottky junction and promotes electron extraction from BiVO$_4$.[118] In order to tune the electronic band structure of BiVO$_4$, Duan *et al.* co-decorated BiVO$_4$ with Ag and Bi nanoparticles.[119] Hollow microstructures of Bi/BiVO$_4$ which was synthesized by solvothermal method showed enhanced performance for the conversion of CO_2 to CO. To further improve photocatalytic reduction towards CO, metallic Ag nanoflakes were loaded on Bi/BiVO$_4$. The increased performance is mainly due to extended visible light response and enhances separation efficiency. Increased attention has been given to copper to substitute noble metals as it also increases the photocatalytic performance as copper acts as a hole/electron mediator which facilitates charge separation and improve the photocatalytic performance.[120] Wei Copi fabricated the composites of BiVO$_4$ and CdS and found increase in the photocatalytic activity towards CO_2 reduction.[121] The 2D-BiVO$_4$ has also been heterojunctioned with perovskites 2D-CsPbBr$_3$ to achieve high efficiency towards CO formation without any co-catalysts or sacrificial agents (Figure 7).[122]

RSC Applied Interfaces Accepted Manuscript
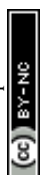
1 The preparation step involved the heterogeneous nucleation of $CsPbBr_3$-sheets on the surface

2 of $BiVO_4$ through annealing step in inert atmosphere, which outlines the stability of $BiVO_4$ as

3 the substrate surface. The presence of oxygen vacancies in $BiVO_4$ achieved a gradient Fermi

4 level shift, and enlarges the Fermi level gap between the semiconductors. This process is

5 accompanied with enhanced interfacial electric field in the heterojunction, which consecutively

6 promoted the charge carrier separation process. It was interesting to note that pure $CsPbBr_3$

7 had a quasi-square morphology with random sizes, while uniform nanosheets were grown on

8 the $BiVO_4$ substrate surface, which further emphasize the pivotal role of substrate surface in

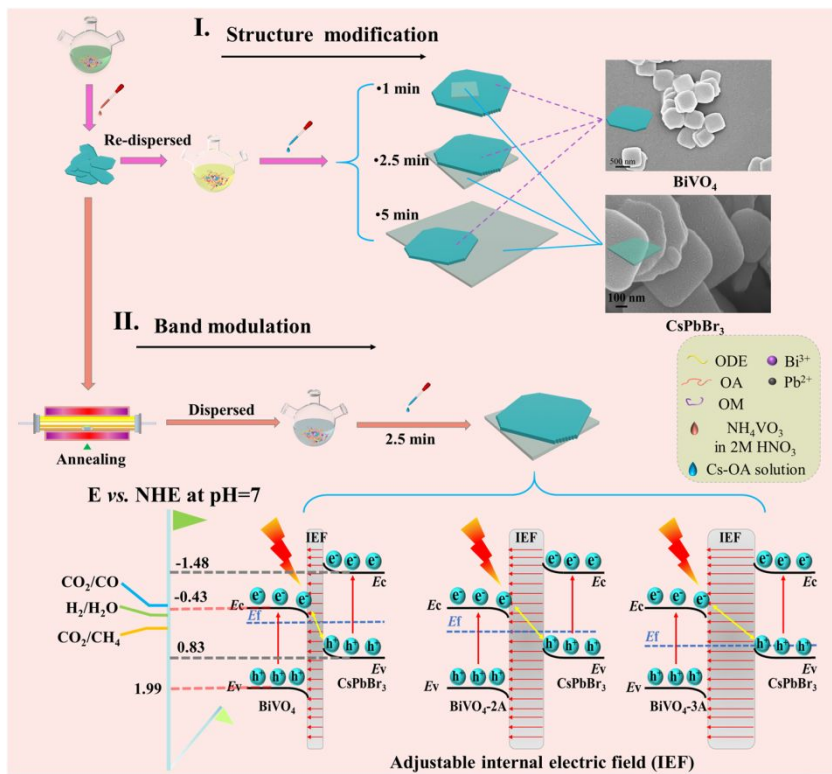9 altering the morphological features during the nucleation process.[122]

22 **Figure 7.** Schematic illustration of the synthesis, structural modification and band modulation
23 of the $BiVO_4/CsPbBr_3$ heterojunction. Reprinted with permission from ref.122. Copyright
24 2022, Elsevier.

26 Wang *et al.* further explored from the synthesis of $Bi_2S_3/BiVO_4$ heterostructure which was

27 obtained from in situ selective ion exchange method. Authors loaded the cocatalyst MnO_x on

1 the heterojunction photocatalysts to improve the selectivity.[60] In this work, photooxidation

2 deposition method was adopted to decorate MnO_x facet selectively. $BiVO_4$ was synthesized by

3 hydrothermal method and Bi_2S_3 nanosheets were preferentially grown on (110) of $BiVO_4$.

4 Bi_2S_3 was grown on (110) of the $BiVO_4$ by sulphurisation. As the concentration of S^{2-} ions

5 increases, surface of $BiVO_4$ turns rough as VO^{3-} ions will be replaced by S^{2-} since the solubility

6 of Bi_2S_3 is more than $BiVO_4$. It is worth to note that if the reaction time is less than 90 minutes,

7 the crystallinity of Bi_2S_3 was poor and structurally unstable. Therefore, effective interfacial

8 contact will be unsuccessful in the short duration of the reaction.[60] However, as the time

9 duration progresses to 120 minutes, the reaction will promote non-selective epitaxial growth.

10 The sponge like structure of MnO_x grows selectively on (110) facets of $BiVO_4$ rather than on

11 (010) facets of Bi_2S_3 nanosheets.[60] Photocatalytic reduction of CO_2 to CH_3OH was observed

12 in this heterostructure with the production rate of 20 ± 2.33 $\mu molg^{-1}h^{-1}$.

13 Type-II heterojunction of $Bi_4Ti_3O_4$ with $BiVO_4$ was fabricated through simple electrospinning

14 technique and solvothermal method.[123] $Bi_4Ti_3O_4$ with $BiVO_4$ formed Type-II heterojunction

15 which yielded various intermediates to give CO and CH_4OH (Figure 8). Direct S-scheme

16 heterojunction was formed between $BiVO_4$ and TiO_2 which changed the charge separation.

17 Platinum decoration was carried out to enhance the CO_2 photoreduction. Platinum decoration

18 on the surface of these catalysts helped in converting CO_2 to CH_4 by mitigating electron hole
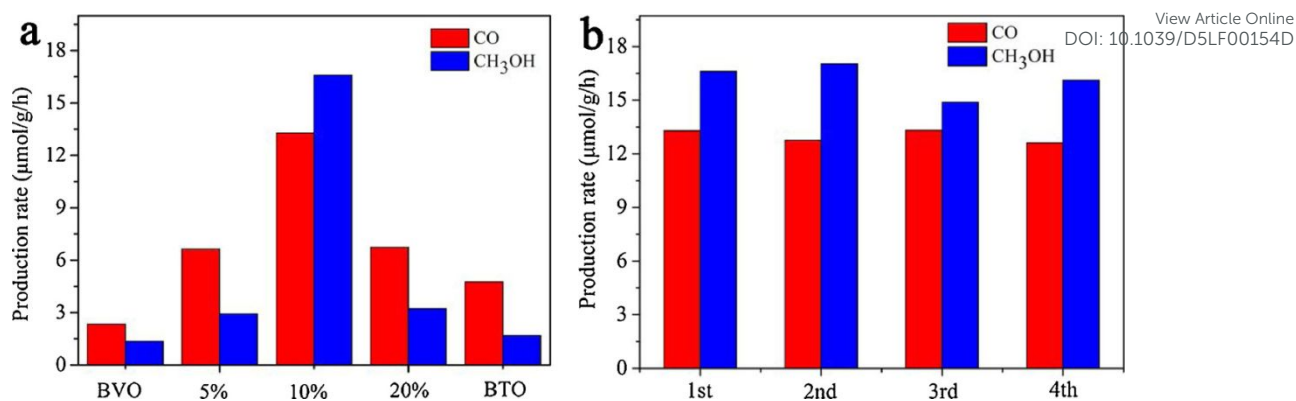
19 pair recombination.[124]

20

21

22

23
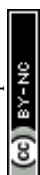
Figure 8. (a) CH_3OH and CO evolutions on the $BiVO_4/Bi_4Ti_3O_{12}$ with different $Bi_4Ti_3O_{12}$ content (b) photocatalytic activity stability of the $BiVO_4/10$ % $Bi_4Ti_3O_{12}$ sample. Reprinted with permission from ref.123. Copyright 2020, Elsevier.

S-scheme heterojunction was fabricated by growing $BiVO_4$ on the surface of a metal organic framework PCN-224 (Cu). This photocatalyst exhibited 100% CO selectivity. Z-scheme heterojunction of $BiVO_4$ and Bi_2S_3 was constructed and MnO_x catalyst was selectively decorated on the surface of (110) facet of $BiVO_4$ by photodeposition technique. Liu *et al.* has decorated Co-Pi, a cocatalyst on $BiVO_4/SnO_2$ and coupled with Au cathode to achieve better conversion efficiency for CO_2.[125] The Z-schematic pathway for charge carriers were constructed using $SrTiO_3$: Rh and $BiVO_4$.[126] In this work, powder of the photocatalyst was used without adjusting the pH. It is worth noting here that the conversion of CO_2 to CO was progressed without any addition of additives.
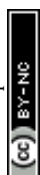
3.3 Bi_2WO_6

Bi_2WO_6 belonging to an Aurivillius family of oxides has been widely researched for its photocatalytic activity owing to its structure, composition and electronic property. Due to the presence of $[Bi_2O_2]$ and perovskite type layers, Bi_2WO_6 possesses narrow band gap which is around 2.6 eV. Due to sluggish reaction kinetics and high rate of recombination of
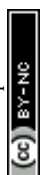
1 photoinduced charge carriers, photocatalytic performance is poor.[127] Therefore, in order to

2 improve the photocatalytic activity, several modifications have been made to Bi_2WO_6 such as

3 morphology control, heterojunction construction and doping.[128] It is important to have an

4 update on the recent works for Bi_2WO_6 to design any new material or to modify the existing

5 photocatalyst. As the band gap of Bi_2WO_6 is wider and unable to utilize the maximum visible

6 light, the heterojunction is constructed with other semiconductors such as $Cs_3Bi_2Br_9$, TiO_2,

7 Cu_2O, $InVO_4$, C_3N_4, BiOBr and BiOI.[129-133]

8 NiO with narrow band gap ~2.5 eV which can efficiently absorb visible light is

9 composited with Bi_2WO_6 which gives a S-scheme charge transfer. The heterojunction yielded

10 CO and CH_4 with 46.9 and 12.1 μmol g^{-1} with CO* intermediate.[134] Lulu Zhao *et al.* attempted

11 to improve the selectivity of the catalyst towards CO from CO_2 by substituting Bi^{3+} and Ag^+

12 ions through liquid phase exchange process.[135] Incorporation of Ag^+ not only improved the

13 adsorption of CO_2 and H_2O but also enhanced the separation of the charge carriers.[135] Ultrathin

14 Bi_2WO_6 nanosheets with Ag^+ and Bi^{3+} exhibits conversion of CO_2 to CO with yield 116 μmol

15 g^{-1} with the selectivity of 95.7 % after 6 h reaction without using any sacrificial agents. The

16 insitu DRIFT suggested that the intermediate formed is COOH* which then converted to

17 thermodynamically stable CO*. It was possible to obtain methane from CO_2 photoreduction

18 from Bi_2WO_6 by modifying the surface with surface plasmon resonance. Precisely controlling

19 the oxygen vacancy in Bi-O-Bi and W-O-W lead to high selectivity towards CH_4. The vacancy

20 created an energy level which is very close to conduction band helped in improving the life

21 span of the charged carriers. To further improve the charge transport pathway, Bi_2WO_6 was

22 covalently heterostructured with Bi_2O_3. Bi_2O_3 nanosheets present on the Bi_2WO_6 surface

23 renders more active sites for which enhances the adsorption of CO_2 on the surface.

24 Graphdiyne (GDY) is explored as a potential material for solar energy harvesting since its first

25 synthesis.[136, 137] Ultrathin heterojunction of GDY and Bi_2WO_6 was prepared by simple

1   hydrothermal method as GDY is an excellent material for efficient charge transfer.[138] Also,

2   increased surface area helped in improving the CO_2 adsoprtion on the photocatalyst.

3   GDY/Bi_2WO_6 yielded f 2.13 and 0.23 µmol h^{-1} g^{-1} of CH_3OH and CH_4 respectively. The

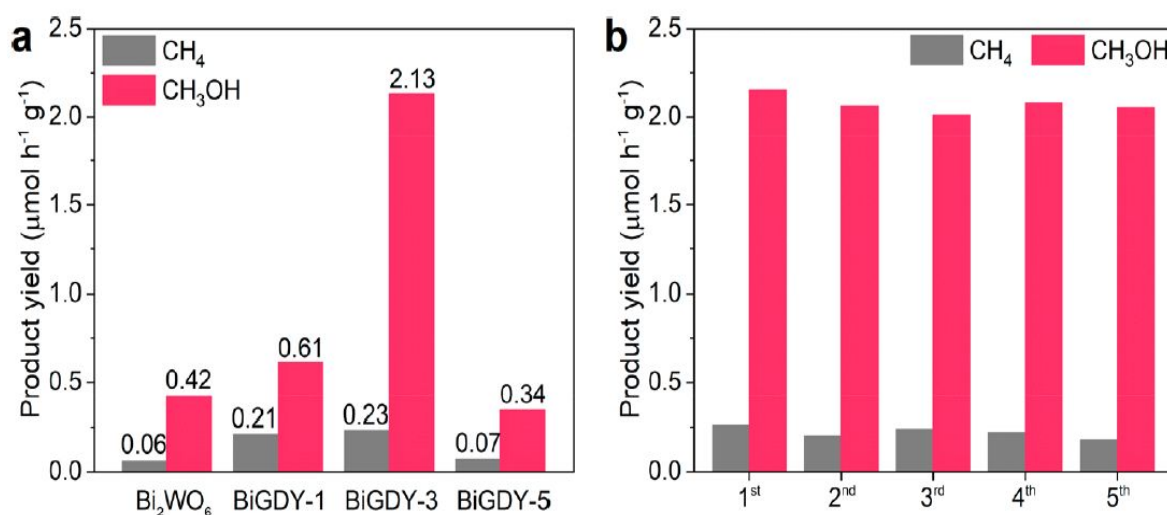4   reduction of CO_2 performance yielded CH_3OH and CH_4 (Figure 9).[138]

Figure 9. (a) Photocatalytic CO_2 reduction performance of samples. (b) Cycling tests of GDY/Bi_2WO_6 heterojunction (BiGDY). Reproduced with permission from ref.138. Copyright 2021, American Chemical Society.

9   Further, to enhance the light absorption ability of the material, hierarchical hollow structured

10 heterojunction of Bi_2WO_6/TiO_2 was constructed by in situ synthetic technique. This

11 heterojunction interface generated the $Bi^{(3-x)+}$ active site which can suppress the charge

12 carriers' recombination very effectively. In producing CO from CO_2 photoreduction, this

13 heterojunction was better than that of pristine Bi_2WO_6 and Bi_2WO_6/TiO_2.[27] Similarly, Bi_2WO_6

14 was heterojunctioned with other semiconductors such as $InVO_4$, $La_2Ti_2O_7$, ZnV_2O_6 either to

15 improve the light absorption ability or to suppress the recombination rate of electron hole
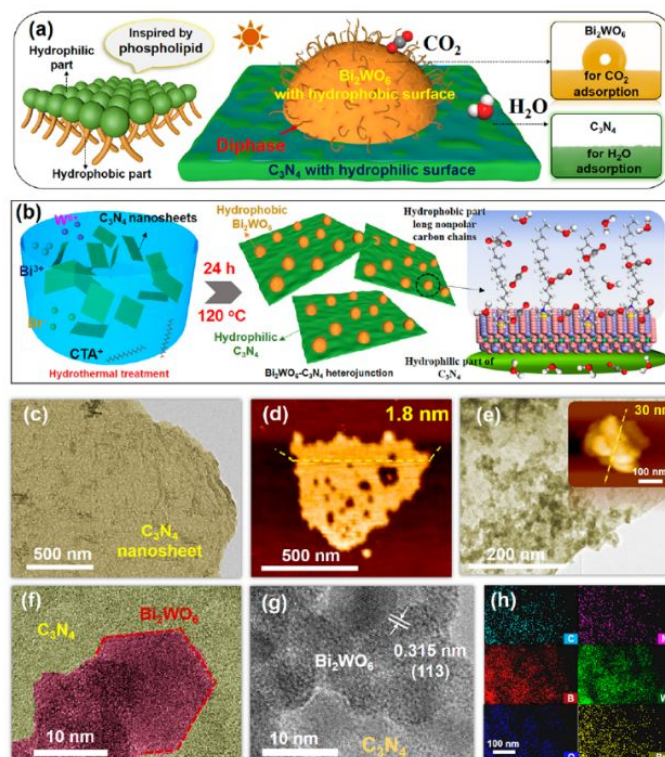
16 pairs.[139-141]

Figure 10. (a) Schematic of the hydrophobic Bi_2WO_6-hydrophilic C_3N_4 photocatalyst and distribution of CO_2 and H_2O. (b) Schematic of the preparation of a phospholipid-mimicking photocatalyst. (c) TEM images and (d) AFM image of C_3N_4. (e) TEM image and AFM image (inset) of Bi_2WO_6-CN. (f and g) HR-TEM images of Bi_2WO_6-C_3N_4. (h) EDS elemental mapping of Bi_2WO_6-C_3N_4. Reproduced with permission from ref. 143. Copyright 2024, American Chemical Society.
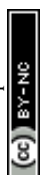
Qiaoya Tang *et al.* constructed a S-scheme heterojunction of C_3N_4/Bi_2WO_6 using electrostatic self-assembly method.[142] Authors claim that the formed ultrathin 2D/2D C_3N_4/Bi_2WO_6 heterostructure offers abundant reaction sites which promoted efficient charge transfer. Yunpeng Liu *et al.* introduced hydrophobic and hydrophilic ends in the diphasic photocatalyst to solve the competitive adsorption issue of CO_2 and H_2O. This amphipathic diphasic hydrophobic and hydrophilic photocatalyst not only inhibits the recombination of
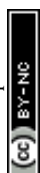
1   charge carriers but also it can enrich CO_2 and H2O on hydrophobic and hydrophilic surfaces

2   respectively as depicted in Figure 10.[143]

3   In an effort to improve the yield of CH_4 in CO_2 photoreduction, Yan-Yang Li *et al.* modified

4   Bi_2WO_6 with chloride ions in view of considering proton produced during the water oxidation

5   affecting the CH_4 generation.[144] Density functional theory confirmed that the mappings of

6   Bi_2WO_6-C_3N_4. chloride ions present on the surface of Bi_2WO_6 nanosheets not only promoted

7   water oxidation but also favoured the formation of CHO* intermediate which facilitated the

8   formation of CH_4. Due to the presence of chloride ions on the surface, the rate of oxidation half

9   reaction goes smoothly which indirectly promotes reduction half reaction (CO_2 reduction).[144]

10   As 2D/2D nanosheets provide large specific surface area and rich active sites, Yong Jiang *et*

11   *al.* fabricated the heterojunction of $Cs_2AgBiBr_6$/Bi_2WO_6.[145] This enabled Z-scheme charge

12   transfer and reduced charge carrier recombination to obtain the yield of 1582.0 $\mu mol\ g^{-1}$ for

13   CO and 8602 $\mu mol\ g^{-1}$ of CH_4.[145]

14   Black phosphorous (BP) is a 2D material which is considered as a potential candidate

15   for photocatalytic applications as it provides short diffusion length for the charge carriers. Since

16   BP has large specific area, it offers ample reaction sites for CO_2 adsorption. Therefore, BP has

17   been used for constructing heterojunction with Bi_2WO_6 as it is believed to accelerate multistep

18   electron process and dimerization of C-C bond. Minghui Zhang *et al.* fabricated 2D/2D

19   BP/Bi_2WO_6 by electrostatic assembly method.[146] Organic oxidation to value added product

20   was coupled with CO_2 reduction which lowers the thermodynamic barrier for water oxidation.

21   In this regard, benzylamine which is a derivative of the renewable biomass was used for organic

22   oxidation to give imines as high value product. The adsorption study revealed that Bi_2WO_6

23   with BP exhibited more active sites for CO_2 adsorption. In addition to this, benzylamine

24   provided alkaline media which increases the solubility of CO_2 in the solution. In situ Fourier

25   transform infrared radiation (FTIR) suggested CO* and CH_3* are the intermediates for the

RSC Applied Interfaces Accepted Manuscript

1 conversion of CO_2 to C_2H_5OH.[135] To harvest visible light absorption and to improve charge

2 transfer, g-C_3N_4 was introduced on Bi_2WO_6. The photocatalytic performance was further

3 enhanced by incorporating reduced graphene oxide.[142, 147] In another report, ultrathin 2D/2D

4 Bi_2WO_6/g-C_3N_4 offered abundant contact interfaces with more accessible reaction sites that

5 exhibited higher selectivity for CO generation compared to CH_4. The higher content of Bi_2WO_6

6 hampered the performance of the composite, as they can shield the incident photons striking

7 the catalyst surface and hinder the bandgap excitation process. Furthermore, composite retained

8 its performance even after four consecutive cycles.[142] These 2D/2D heterojunction with face-

9 to-face interfaces can lower the intrinsic resistance and boost the charge carrier separation

10 process.[148]

11   The deposition of QDs like $Cs_3Bi_2Br_9$ on the Bi_2WO_6-nanosheets can selectively

12 promote the CO_2 reduction reactions to form CO under the visible light.[133] This unique 0D/2D

13 (dot-to-face) geometry and lead free bismuth halide perovskite semiconductor remained as an

14 added advantage for this composite. The larger surface area of 2D sheets enabled the finer

15 distribution of QDs on its surface, which not only resulted in stable heterojunction but also

16 lowered the distance for charge carrier migration process. The preparation method was flexible

17 as it only involved the dispersion of positively charged QDs on the negatively charged sheets

18 isopropanol suspension through electrostatic self-assembly approach. This strategy has benefit

19 from the prospect of retaining the pristine morphology of concerned semiconductors during the

20 heterojunction formation. The ESR analysis revealed the generation of both hydroxyl and

21 superoxide radicals, which further confirms the formation of S-scheme heterojunction between

22 them.[133]

23 **3.4 Bi_2MoO_6**

24   Bi_2MoO_6 is also found to be an interesting material which belongs to Aurivillius oxide

25 family. The structure of Bi_2MoO_6 consists of alternating array of [Bi_2O_2] and [MoO_6] octahedra
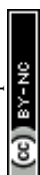
1   layers. This ternary metal oxide is an *n*-type semiconductor possessing narrow band gap, which

2   can utilize visible light of the solar spectrum.[149, 150] It is possible to effectively tune the

3   electronic distribution and morphology to optimize the photocatalytic application towards CO_2

4   reduction. By changing the temperature, reaction time and solvent, it is easier to modify the

5   morphology of Bi_2MoO_6. Therefore, enormous effort has been put to improve the

6   photocatalytic performance of Bi_2MoO_6 by morphology control and heterojunction composite.

7   Dai *et al.* has grown Bi_2MoO_6 quantum dots on reduced graphene oxide scaffolds using

8   economically viable hydrothermal method.[151] Due to high surface area of reduced graphene

9   oxide, dispersion of Bi_2MoO_6 quantum dots was enhanced. The electron migration from

10 Bi_2MoO_6 quantum dots to reduced graphene oxide was facile as reduced graphene oxide

11 exhibits excellent electrical conductivity. The photocatalytic mechanism is illustrated in Figure
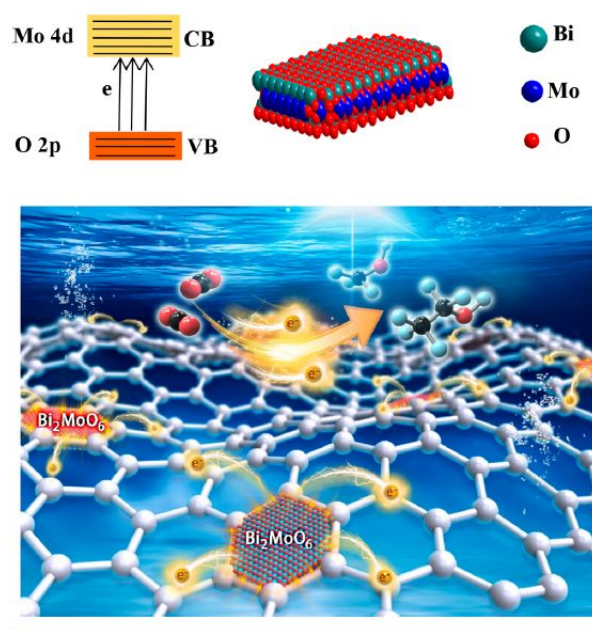
12 11.

23 **Figure 11.** Unit cell model of Bi_2MoO_6, and electron Transfer. Reproduced with permission.

24 Reprinted with permission from ref.151. Copyright 2020, American Chemical Society.

Yu *et al.* has fabricated hierarchical heterostructure of 2D nanosheets of Bi_2MoO_6 was grown on 1D In_2S_3 using solvothermal technique. Hollow nanotubes of In_2S_3 nanostructure were designed using MIL-68 precursor. The surface oxygen vacancy was created by using N,N,N',N'-Tetramethylethylenediamine (TMEDA). However, the morphology of Bi_2MoO_6 nanosheets get distorted to give particle like morphology if TMEDA is used to introduce surface oxygen vacancy.[152] The presence of interfacial Mo-S bond in Bi_2MoO_6/$ZnIn_2S_4$ boosted the production of CO from CO_2 reduction reactions.[150] The in-situ loading of $ZnIn_2S_4$ nanoflakes on the porous microspheres of Bi_2MoO_6 under the solvothermal conditions, offered plentiful catalytic sites for CO_2 adsorption and activation process. Furthermore, the hierarchical structures enhanced the visible light absorption and also possessed good photostability and reusability. The generation of superoxide and hydroxyl radicals as confirmed from the ESR analysis confirmed the formation of S-scheme heterojunction between them.[150] The oxygen vacancies engineered Bi_2MoO_{6-x}/MoS_2 was active for CO_2 reduction to form CO, wherein oxygen vacancies were obtained from pristine Bi_2MoO_6 through simple annealing step. It was interesting to note that the oxygen vacancies did not disrupt the microsphere-like morphology of the host matrix and density of oxygen vacancies increased upon composite formation with MoS_2-nanosheets. The presence of these oxygen vacancies lowered the bandgap of host matrix and act as effective electron pedals to delay the accumulation of charges and also reduces the work function of host matrix, which indirectly lessens the energy required for the electron transfer process.[153]
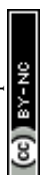
Maryam Ahmadi *et al.* heterojunctioned with the benchmark photocatalyst TiO_2 to reduce the rate of recombination of photogenerated electron-hole pair.[154] Synergistic effect was achieved by compositing Bi_2MoO_6 with TiO_2 as the band gap of both materials achieve staggered configuration. Bi_2MoO_6 nanosheets were decorated TiO_2 nanobelts by simple solvothermal method. As cocatalysts act as electron scavengers, Pt-Cu was decorated on the composite by
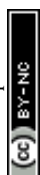
1 reduction method. It was noticed that the photo response of the photocatalyst has increased and

2 obtained 34.6 μmol g^{-1} of methane for Bi_2MoO_6/TiO_2. Once the Pt-Cu were loaded on the

3 composite, the methane production was increased further as cocatalyst help in trapping the

4 electrons and reduce recombination rate.[154]

5 Other metal oxide semiconductor materials have also been heterojunctioned such as a spinal

6 compound $ZnFe_2O_4$ which produced CO and CH_4.[155] The S-scheme pathway helped in slight

7 improvement of the CO_2 reduction; the catalyst needs to be improved to achieve better

8 photocatalytic performance. The 2D/2D $Bi_2MoO_6/Zn_3V_2O_8$ which formed S-scheme produced

9 CO and CH_4.[156] The Ni containing metal organic framework (MOF) on Bi_2MoO_6 exhibits great

10 potential towards photocatalytic CO_2 reduction. Similarly, Co-MOF has also been synthesized

11 and found that the photocatalyst yields CO and CH_4.

12 Nanosheets, belts, particles and spheres of Bi_2MoO_6 have been synthesized, reports on quantum

13 dots of Bi_2MoO_6 is uncommon.[157-159] Weili Dai *et al*. anchored quantum dots of Bi_2MoO_6 on

14 2D reduced graphene oxide (rGO) sheets which act as an electron reservoir helps in boosting

15 the CO_2 reduction. Quantum dots $Bi_2MoO_6/2D$ rGO produced 84.8 μmol g^{-1} of methanol and

16 57.5 μmol g^{-1} of ethanol which is highest among bismuth based photocatalyst towards

17 reduction of CO_2. [151] Further, the quantum dots of Bi_2MoO_6 was deposited on the pomelo peel

18 derived carbon (CPP) to improve the photocatalytic property of the catalyst. The CPP possesses

19 high specific surface area and has excellent thermal conductivity. As the photothermal

20 temperature increases, it not only enhances the migration rate of the electrons and holes but

21 also promotes adsorption of CO_2. Mingnv Guo *et al.* synthesized 0D/3D CPP-A/Bi_2MoO_6 to

22 achieve photothermal synergy to effectively utilize the solar spectrum.[160] As temperature

23 increases up to 100 °C, migration ability is increased and C-O-Bi bridge is formed which

24 enhanced the electron transfer at the interface. This work would provide a solution for bottle

25 neck problem of poor charge carrier transport in photothermal catalytic activity.[160]

1 Combining both n-type semiconductors In_2S_3 and Bi_2MoO_6, S-scheme was achieved to design

2 an efficient photocatalyst. 2D nanosheets of Bi_2MoO_6 was grown on 1D nanotubes of In_2S_3

3 reduced the carrier diffusion length. Z-scheme heteorjunctions such as Bi_2MoO_6/CdS,

4 Bi_2MoO_6/CeO_2 have also been constructed to improve the charge carrier transport.[161, 162] S-

5 Scheme heterostructure can also be achieved from $ZnIn_2S_4$ decorated on Bi_2MoO_6 where Mo-

6 S bond at the interface helped in improving the photocatalytic activity.[150] S-scheme

7 heterostructures can also be achieved by chalcogenides, especially with MnS.

8 Diethylenetriamine (DETA) ammonia treated MnS hollow sphere heterostructured with Bi

9 surface plasmon resonance enhanced CO_2 reduction rate of Bi_2MoO_6. The presence of basic

10 functional groups on DETA reduced the activation energy of acidic CO_2 whereas surface

11 plasmon effect from Bi produces hot electrons and decreases the recombination rate of charge

12 carriers. This combined effect helped in boosting the CO_2 photocatalytic reduction reaction.

13

14 **Table 1.** Comparison of heterojunction of bismuth-based materials and their photocatalytic

15 conversion of CO_2 along with synthesis technique.

16

Bismuth based material	Preparation Method	Nature of hetero-junction	Products: Production Rate in $\mu molg^{-1}h^{-1}$	Reference
BiOCl/BiOBr	Hydrothermal Temperature - 160 ℃ Time - 12 h	Type-I	CO: 7.353	[84]
Bi_2WO_6/TiO_2	Solvothermal	Type-II	CO: 43.7	[27]
$BiOCl/Bi_2WO_6$	Hydrothermal Temperature - 120 ℃ Time - 12 h	Type-II	CO: 6.63	[76]
$BiOBr/Bi_2S_3$	Hydrothermal Temperature - 180 ℃	Type-II	CO: 103,500	[92]

Material	Method	Type	Products	Ref
	Time - 2 h			
BiOBr/CoAl LDH	Hydrothermal Temperature - 160 °C	Type-II	CO: 4.096 CH_4: 4.174	[96]
	Time - 12 h			
BiOI/In_2O_3	Hydrothermal method Temperature - 180 °C	Type-II	CO: 11.98 CH_4: 5.69	[107]
	Time - 12 h			
$BiVO_4$/$Bi_4Ti_3O_{12}$	Hydrothermal Temperature - 160 °C	Type-II	CH_3OH: 16.6 CO:13.29	[123]
	Time - 21 h			
$BiVO_4$/Bi_2S_3/MnO_3	Selective epitaxial growth Temperature - 180 °C	Z-scheme	CH_3OH: 20 ± 2.33	[60]
	Time – 90 mins			
BiOBr/HNb_3O_8	Self-Assembly ultrasonic dispersion for 30 min	Z-Scheme	CO: 164.6	[163]
g-C_3N_4/BiOI/RGO	hydrothermal method Temperature - 160 °C	Z-Scheme	CO: 21.85	[164]
	Time - 6 h			
Co-MOF/Bi_2MoO_6	Solvothermal followed by insitu growth method Temperature - 165 °C	Z-scheme	CO: 19.76 CH_4: 8.24	[165]
	Time - 16 h			
$BiVO_4$/Cu_2O/Bi	Solvothermal Temperature - 120 °C	Z-Scheme	CH_4: 1.8 CO: 8.4	[166]
	Time - 6 h			
In_2O_3/Bi_2S_3	Hydrothermal followed by ultrasonication Temperature - 180 °C	Z-scheme	CO: 2.67	[167]
	Time – 24 h			

quinacridone/BiVO$_4$	self-assembly method Stirring for 1 h	Z-Scheme	CO: 407 CH$_4$: 29	[168]
BiOCl/C$_3$N$_4$	Hydrothermal Temperature - 120 °C Time – 12 h	Z-scheme	CO: 45.33	[169]
BiOBr-(001)/Bi$_2$SiO$_5$/Bi	Solvothermal method Temperature - 190 °C Time – 12 h	S-Scheme	CO: 234.05	[74]
BiOBr/Bi$_2$WO$_6$	Hydrothermal Temperature - 160 °C Time – 6 h	S-Scheme	CO: 55.17	[94]
BiOBr/Cu$_2$O	Hydrothermal Temperature - 160 °C Time – 12 h	S-scheme	CH$_4$: 22.78	[97]
BiOBr/CsPbBr$_3$	Self -assembly process Overnight magnetic stirring in dark.	S-scheme	CO: 104.4 CH$_4$: 10.0	[103]
BiVO$_4$/CsPbBr$_3$	In situ colloidal growth method	S-scheme	CO: 103.5	[122]
Cs$_3$Bi$_2$Br$_9$/Porous BiOCl	Dipping method Magnetic stirring for 1 h.	S-scheme	CO: 25.5	[170]
BiVO$_4$/Cu-Bi	Solvothermal Temperature - 120 °C Time – 6 h	Oxygen vacancy	CO: 2.96	[56]
BiOCl/Br-O$_V$	Solvothermal Temperature - 160 °C Time – 10 h	Defects	CO: 7.37	[171]

1   Note: Readers are requested to follow the respective references for more information.

2   Table 1 illustrates different interfacial engineering of bismuth-based materials with various

3   other compounds and their respective CO$_2$ conversion product. The synthesis method and also

1 charge carrier pathway followed by the heterojunction is also mentioned. It is interesting to

2 note that most of the bismuth-based materials are fabricated by wet-chemical approaches such

3 as hydrothermal and solvothermal methods. However, these techniques pose several

4 challenges, as the materials purity and morphology are sensitive to synthesis temperature and

5 reaction time. Therefore, reproducibility of the structure-morphological features would be

6 difficult for the large-scale synthesis.[172, 173]

7   Other bismuth containing materials such as $Bi_{12}O_{17}Br_2$/g-C_3N_4, $Bi_{12}O_{17}Cl_2$/g-C_3N_4,

8 Bi_3NbO_7/g-C_3N_4, $SrBi_4Ti_4O_{15}$/Bi_2O_3, Cs_2CuBr_4/Bi_2O_3, $BiOIO_3$/CdS and Bi/$Bi_4V_2O_{11}$, are

9 explored for CO_2 photocatalytic conversion.[174-181] In particular, the presence of interfacial Bi-

10 N bonds and the nitrogen vacancy in $Bi_{12}O_{17}Br_2$/g-C_3N_4 achieved simultaneous tetracycline

11 degradation and CO_2 reduction reactions.[175] In fact, photocatalytic reduction of CO_2 was

12 improved in the presence of tetracycline compared to Rhodamine-B and phenol molecules,

13 which was attributed to the superior electron donating ability of TC. The flower-like $Bi_{12}O_{17}Br_2$

14 were decorated on the defective g-C_3N_4 surface under the wet-chemical approach. The presence

15 of defects was important to drive the coupled photocatalytic reactions. It was proposed that

16 *CO_2 was transformed to *COOH through hydrogenation step, which upon further protonation

17 yielded *CO which was feasible for the desorption from the composite surface. This work is a

18 classical illustration of win-to-win strategy achieve both environmental and energy related

19 issues.[175] Alternatively, $Bi_{12}O_{17}Cl_2$/g-C_3N_4 promoted the formation of CH_4 through two-

20 electron and two-proton reaction pathways.[179] On the other hand, Bi_3NbO_7/g-C_3N_4 promoted

21 the formation of CH_4 as the reduction pathway proceeded through the formation of CH_3O* and

22 CHO* fragments.[176] Lee *et al.* reported CdS-nanorods/$BiOIO_3$-nanosheets (1D/2D) selectively

23 generated CO and minor amounts of H_2 during the CO_2 reduction reactions and activity was

24 retained even after four consecutive cycles.[180] The line interfacial contact between the

25 integrated materials promoted the charge carrier lifetime with minimal recombination

RSC Applied Interfaces Accepted Manuscript
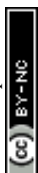
1 pathways.[182] Significant improvement in photocatalytic performance was noticed, yet these

2 materials must be explored further to enhance the conversion of CO_2 to value added product.

3 **Conclusion and perspective**

4 Photcatalytic reduction of CO_2 into the fuels and simple chemicals is a versatile

5 approach that can simultaneously address the environmental issues, carbon neutrality and the

6 energy demands. The high degree of chemical inertness of CO_2, sluggish dynamics of multi-

7 electron reduction reactions and massive charge carrier recombination in single-phase

8 photocatalytic active materials has driven the emergence of heterojunction photocatalysts. In

9 this context, bismuth based materials found to be interesting class of material as they possess

10 narrow band gap and favourable electronic structure for photocatalytic CO_2 reduction reaction.

11 Although notable progress is achieved with Bi-based heterojunctions, the achieved efficiency

12 is still far from the satisfactory, as the presence of single electric field may not always boost

13 the charge carrier separation process. Thus, future research must essentially focus on the

14 coupling of multiple electric fields, which not only extend the carrier life time, but also allows

15 their diffusion to react with the adsorbed species. Furthermore, design and fabricating the

16 hierarchical Bi-based heterojunctions comprising QDs, 1D, 2D and 3D forms would be even

17 more advantageous from the prospect of improving the light harvesting capacity, surface

18 reactions and charge carrier separation kinetics. In addition, some importance should be given

19 for the preparation of NIR responsive Bi-based materials, which would spotlight our interest

20 to harvest the complete solar light for the photocatalytic reactions. From the viewpoint of

21 charge carrier dynamics, steering the migration of charge carriers in a specific path without

22 compromising their lifetimes requires a suitable interface during the fabrication of

23 heterojunctions. Alongside, coupling of CO_2 reduction reaction with pollutant degradation,

24 H_2O_2 generation and synthesis of organic compounds will full-fill our dream of achieving the

25 win-to-win strategy comprising energy and environmental aspects.
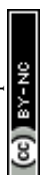
1 This review summarizes the interfacial engineering carried out for bismuth-based materials in

2 recent years for CO_2 reduction. A large number of reported works have demonstrated that the

3 interfacial engineering helps to effectively enhance the charge separation, light absorption and

4 adsorption of CO_2 on the surface of the photocatalyst. However, the conversion efficiency is

5 far from efficiency required to implement in practical applications. Most of the interfacially

6 engineered bismuth based photocatalysts exhibited the rate of reduction of CO_2 less than 100

7 μmol g^{-1}. It is also noticed from the previous literature that the product obtained from the

8 photocatalytic reduction reaction is carbon monoxide. Though $BiVO_4$ shows promising results

9 in photoelectrochemical water splitting, photocatalytic activity towards CO_2 reduction is

10 insignificant even after interfacial engineering with several other competent materials.

11 Therefore, attention must be paid to enhance the photocatalytic activity of bismuth-based

12 materials to meet the practical requirements for photocatalytic CO_2 reduction.
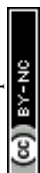
13

14 **Declaration of the competing interest**

15 Authors declare that there is no competing financial interest.

16

17 **Data Availability**

18 No data used in this article.

19

20 **References**

21 1.   B. Kumar, M. Llorente, J. Froehlich, T. Dang, A. Sathrum and C. P. Kubiak, *Annual*
22   *Review of Physical Chemistry*, 2012, **63**, 541-569.
23 2.   T. Abbas, H. S. M. Yahya and N. A. S. Amin, *Energy & Fuels*, 2023, **37**, 18330-18368.
24 3.   R. Pang, K. Teramura, M. Morishita, H. Asakura, S. Hosokawa and T. Tanaka,
25   *Communications Chemistry*, 2020, **3**, 137.
26 4.   S. J. Cobb, S. Rodríguez-Jiménez and E. Reisner, *Angewandte Chemie International*
27   *Edition*, 2024, **63**, e202310547.
28 5.   U. Ulmer, T. Dingle, P. N. Duchesne, R. H. Morris, A. Tavasoli, T. Wood and G. A.
29   Ozin, *Nature Communications*, 2019, **10**, 3169.
30 6.   H. S. Shafaat and J. Y. Yang, *Nature Catalysis*, 2021, **4**, 928-933.

RSC Applied Interfaces Accepted Manuscript

RSC Applied Interfaces Accepted Manuscript

7.   G. Gadikota, *Commun Chem*, 2021, **4**, 23.

8.   B. Wang, S. Chen, Z. Zhang and D. Wang, *SmartMat*, 2022, **3**, 84-110.

9.   D. H. Apaydin, *Israel Journal of Chemistry*, 2022, **62**, e202100085.

10.   C. Zuo, Q. Su and X. Yan, *Journal*, 2023, **11**.

11.   S. S. Bhosale, A. K. Kharade, E. Jokar, A. Fathi, S.-m. Chang and E. W.-G. Diau, *Journal of the American Chemical Society*, 2019, **141**, 20434-20442.

12.   T. Zhao, J. Xiong, W. Li, G. Cheng and S. X. Dou, *Chemical Engineering Journal*, 2025, **506**, 160313.

13.   D. K. Chauhan, N. Sharma and K. Kailasam, *Materials Advances*, 2022, **3**, 5274-5298.

14.   S. C. Shit, I. Shown, R. Paul, K.-H. Chen, J. Mondal and L.-C. Chen, *Nanoscale*, 2020, **12**, 23301-23332.

15.   E. Gong, S. Ali, C. B. Hiragond, H. S. Kim, N. S. Powar, D. Kim, H. Kim and S.-I. In, *Energy & Environmental Science*, 2022, **15**, 880-937.

16.   T. Zhao, W. Zhang, J. Xiong, W. Li and G. Cheng, *Separation and Purification Technology*, 2025, **360**, 130849.

17.   G. Yang, J. Xiong, M. Lu, W. Wang, W. Li, Z. Wen, S. Li, W. Li, R. Chen and G. Cheng, *Journal of Colloid and Interface Science*, 2022, **624**, 348-361.

18.   W. A. Thompson, E. Sanchez Fernandez and M. M. Maroto-Valer, *ACS Sustainable Chemistry & Engineering*, 2020, **8**, 4677-4692.

19.   S. G. Kumar, R. Kavitha and C. Manjunatha, *Energy & Fuels*, 2023, **37**, 14421-14472.

20.   B. Tang and F.-X. Xiao, *ACS Catalysis*, 2022, **12**, 9023-9057.

21.   W. Huang, Q. Zhu, Y. Zhu, C. Chen and J. Shen, *Materials Today Energy*, 2023, **38**, 101458.

22.   Z. Dong, B. Li, Y. Zhu and W. Guo, *EES Catalysis*, 2024, **2**, 448-474.

23.   L. D. M. Torquato, F. A. C. Pastrian, J. A. L. Perini, K. Irikura, A. P. de L. Batista, A. G. S. de Oliveira-Filho, S. I. Córdoba de Torresi and M. V. B. Zanoni, *Applied Catalysis B: Environmental*, 2020, **261**, 118221.

24.   S. Wang, G. Wang, T. Wu, Y. Zhang, F. Zhan, Y. Wang, J. Wang, Y. Fu and J. Qiu, *Journal of Materials Chemistry A*, 2018, **6**, 14644-14650.

25.   K. R. Rao, S. Pishgar, J. Strain, B. Kumar, V. Atla, S. Kumari and Joshua M. Spurgeon, *Journal of Materials Chemistry A*, 2018, **6**, 1736-1742.

26.   Y. Xiao, A. Abulizi, K. Okitsu and T. Ren, *Journal of Industrial and Engineering Chemistry*, 2023, **125**, 317-324.

27.   W.-R. Liu, S. Yu, Z. Liu, P. Jiang, K. Wang, H.-Y. Du, Z.-Y. Hu, M.-H. Sun, Y.-L. Wang, Y. Li, L.-H. Chen and B.-L. Su, *Inorganic Chemistry*, 2024, **63**, 6714-6722.

28.   Y. Ma, X. Yi, S. Wang, T. Li, B. Tan, C. Chen, T. Majima, E. R. Waclawik, H. Zhu and J. Wang, *Nature Communications*, 2022, **13**, 1400.

29.   A. Sharma, A. Hosseini-Bandegharaei, N. Kumar, S. Kumar and K. Kumari, *Journal of CO2 Utilization*, 2022, **65**, 102205.

30.   J. Wu, Y. Huang, W. Ye and Y. Li, *Advanced Science*, 2017, **4**, 1700194.

31.   S. G. Kumar and L. G. Devi, *The Journal of Physical Chemistry A*, 2011, **115**, 13211-13241.

32.   X. Li, Y. Sun, T. Xiong, G. Jiang, Y. Zhang, Z. Wu and F. Dong, *Journal of Catalysis*, 2017, **352**, 102-112.

33.   S. M. Ghoreishian, K. S. Ranjith, H. Lee, H.-i. Ju, S. Zeinali Nikoo, Y.-K. Han and Y. S. Huh, *Journal of Hazardous Materials*, 2020, **391**, 122249.

34.   S. M. Ghoreishian, K. S. Ranjith, B. Park, S.-K. Hwang, R. Hosseini, R. Behjatmanesh-Ardakani, S. M. Pourmortazavi, H. U. Lee, B. Son, S. Mirsadeghi, Y.-K. Han and Y. S. Huh, *Chemical Engineering Journal*, 2021, **419**, 129530.
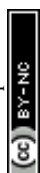
35. S. M. Ghoreishian, K. S. Ranjith, M. Ghasemi, B. Park, S.-K. Hwang, N. Irannejad, M. Norouzi, S. Y. Park, R. Behjatmanesh-Ardakani, S. M. Pourmortazavi, S. Mirsadeghi, Y.-K. Han and Y. S. Huh, *Chemical Engineering Journal*, 2023, **452**, 139435.

36. K. Sridharan, S. Shenoy, S. G. Kumar, C. Terashima, A. Fujishima and S. Pitchaimuthu, *Journal*, 2021, **11**.

37. S. Chen, D. Huang, M. Cheng, L. Lei, Y. Chen, C. Zhou, R. Deng and B. Li, *Journal of Materials Chemistry A*, 2021, **9**, 196-233.

38. Y. Zhang, G. Zhang, J. Di and J. Xia, *Current Opinion in Green and Sustainable Chemistry*, 2023, **39**, 100718.

39. R. He, D. Xu, B. Cheng, J. Yu and W. Ho, *Nanoscale Horizons*, 2018, **3**, 464-504.

40. J. Z. Hassan, A. Raza, U. Qumar and G. Li, *Sustainable Materials and Technologies*, 2022, **33**, e00478.

41. S. S. M. Bhat and H. W. Jang, *ChemSusChem*, 2017, **10**, 3001-3018.

42. K. Xu, L. Wang, X. Xu, S. X. Dou, W. Hao and Y. Du, *Energy Storage Materials*, 2019, **19**, 446-463.

43. D. Chen, Z. Xie, Y. Tong and Y. Huang, *Energy & Fuels*, 2022, **36**, 9932-9949.

44. S. Wang, L. Wang and W. Huang, *Journal of Materials Chemistry A*, 2020, **8**, 24307-24352.

45. A. Kumar, P. Singh, A. A. P. Khan, Q. V. Le, V.-H. Nguyen, S. Thakur and P. Raizada, *Chemical Engineering Journal*, 2022, **439**, 135563.

46. H. Sudrajat and M. Nobatova, *RSC Applied Interfaces*, 2025, **2**, 599-619.

47. J. Low, J. Yu, M. Jaroniec, S. Wageh and A. A. Al-Ghamdi, *Advanced Materials*, 2017, **29**, 1601694.

48. P. C P, M. Srinivas and S. G. Kumar, *Inorganic Chemistry Frontiers*, 2025, **12**, 2138-2181.

49. S. G. Kumar and K. S. R. K. Rao, *RSC Advances*, 2015, **5**, 3306-3351.

50. A. Balapure, J. Ray Dutta and R. Ganesan, *RSC Applied Interfaces*, 2024, **1**, 43-69.

51. S. Harrison and M. Hayne, *Scientific Reports*, 2017, **7**, 11638.

52. W. Zhang, A. R. Mohamed and W.-J. Ong, *Angewandte Chemie International Edition*, 2020, **59**, 22894-22915.

53. A. Ghosh, A. Pramanik, S. Pal and P. Sarkar, *The Journal of Physical Chemistry Letters*, 2024, **15**, 6841-6851.

54. Q. Xu, L. Zhang, B. Cheng, J. Fan and J. Yu, *Chem*, 2020, **6**, 1543-1559.

55. R. Kavitha, C. Manjunatha, J. Yu and S. G. Kumar, *EnergyChem*, 2025, **7**, 100159.

56. L. Huang, Z. Duan, Y. Song, Q. Li and L. Chen, *ACS Applied Nano Materials*, 2021, **4**, 3576-3585.

57. T. Wei, H. Xiao, R. Niu, B. Xu, H. Li, X. Yan, Y. Li, H. Ou, B. Lin and G. Yang, *Chemical Engineering Science*, 2024, **292**, 120012.

58. K. Wang, T. Sun, H. Ma, R.-J. You, Z.-H. He, J.-G. Chen, H. Wang, W. Wang, Y. Yang, L. Wang and Z.-T. Liu, *Separation and Purification Technology*, 2024, **340**, 126786.

59. X. Li, J. Yu, M. Jaroniec and X. Chen, *Chemical Reviews*, 2019, **119**, 3962-4179.

60. M. Wang, S. Zeng, A. R. Woldu and L. Hu, *Nano Energy*, 2022, **104**, 107925.

61. A. E. Mesoudy, D. Machon, A. Ruediger, A. Jaouad, F. Alibart, S. Ecoffey and D. Drouin, *Thin Solid Films*, 2023, **769**, 139737.

62. C. Sushma and S. G. Kumar, *Inorganic Chemistry Frontiers*, 2017, **4**, 1250-1267.

63. P. Qiu, H. Liu, G. Wang, C. Wang, Z. Huang, C. Huang, Y. Lai, C. Xing and K. Liang, *Journal of Water Process Engineering*, 2024, **68**, 106372.

64. S. Liu, J. Sun, G. Ren and X. Meng, *Materials Science in Semiconductor Processing*, 2022, **137**, 106230.

65. S. G. Kumar and K. S. R. K. Rao, *Applied Surface Science*, 2017, **391**, 124-148.

66. Z. Ali, J. Ma, M. Hong and R. Sun, *Journal of Materials Chemistry A*, 2023, **11**, 3297-3314.

67. S.-S. Wang, X. Liang, Y.-K. Lv, Y.-Y. Li, R.-H. Zhou, H.-C. Yao and Z.-J. Li, *ACS Applied Energy Materials*, 2022, **5**, 1149-1158.

68. Y. Wang, E. Chen and J. Tang, *ACS Catalysis*, 2022, **12**, 7300-7316.

69. F. Wang, J. Guo, L. Han, H. Shen, L. Zhu and S. Chen, *Chemical Engineering Journal*, 2023, **478**, 147365.

70. Y. Shi, G. Zhan, H. Li, X. Wang, X. Liu, L. Shi, K. Wei, C. Ling, Z. Li, H. Wang, C. Mao, X. Liu and L. Zhang, *Advanced Materials*, 2021, **33**, 2100143.

71. Y. Wang, H. Wang, L. Guo and T. He, *Journal of Colloid and Interface Science*, 2023, **648**, 889-897.

72. S. Vinoth, W.-J. Ong and A. Pandikumar, *Coordination Chemistry Reviews*, 2022, **464**, 214541.

73. T. L. Yusuf, B. O. Orimolade, D. Masekela, K. A. Adegoke, K. D. Modibane and S. S. Makgato, *Materials Today Sustainability*, 2025, **30**, 101115.

74. X. Guan, X. Zhang, C. Zhang, R. Li, J. Liu, Y. Wang, Y. Wang, C. Fan and Z. Li, *Journal of Colloid and Interface Science*, 2023, **644**, 426-436.

75. H. Jiang, W. Wang, L. Sun, T. Kong, Z. Lu, H. Tang, L. Wang and Q. Liu, *Journal of Catalysis*, 2022, **416**, 1-10.

76. X. Zhao, Y. Xia, X. Wang, N. Wen, H. Li, X. Jiao and D. Chen, *Chemical Engineering Journal*, 2022, **449**, 137874.

77. X. Liu, H. Zhang, X. Qiu, H. Ye, Y. Xie and Y. Ling, *Applied Catalysis A: General*, 2024, **671**, 119574.

78. H. Maimaitizi, A. Abulizi, D. Talifu and Y. Tursun, *Advanced Powder Technology*, 2022, **33**, 103562.

79. W. Li, Y. Zhang, W. Ran, Y. Wang, F. Tian, F. Zhang, M. Xu, D. Zhang, N. Li and T. Yan, *Applied Catalysis B: Environment and Energy*, 2024, **351**, 123978.

80. Z. Huang, J. Wu, M. Ma, J. Wang, S. Wu, X. Hu, C. Yuan and Y. Zhou, *New Journal of Chemistry*, 2022, **46**, 16889-16898.

81. Q. Liu, C. Bai, C. Zhu, W. Guo, G. Li, S. Guo, D. Kripalani, K. Zhou and R. Chen, *Advanced Science*, 2024, **11**, 2400934.

82. Q. Chen, S. Wang, B. Miao and Q. Chen, *Journal of Colloid and Interface Science*, 2024, **663**, 1005-1018.

83. T. Xu, X. Su, Y. Zhu, S. Khan, D.-L. Chen, C. Guo, J. Ning, Y. Zhong and Y. Hu, *Journal of Colloid and Interface Science*, 2023, **629**, 1027-1038.

84. Y. Wu, M. Xu, Y. Wang, B. Hu, Y. Xie and Y. Ling, *Applied Catalysis A: General*, 2024, **677**, 119708.

85. Y. Jiang, Y. Wang, Z. Zhang, Z. Dong and J. Xu, *Inorganic Chemistry*, 2022, **61**, 10557-10566.

86. Y. Chen, F. Wang, Y. Cao, F. Zhang, Y. Zou, Z. Huang, L. Ye and Y. Zhou, *ACS Applied Energy Materials*, 2020, **3**, 4610-4618.

87. Q.-s. Wang, Y.-c. Yuan, X. Hu, D.-w. Jin, F.-k. Zhang, X.-j. Liu, J.-f. Deng, W.-g. Pan and R.-t. Guo, *Energy & Fuels*, 2024, **38**, 4554-4565.

88. A. Nawaz, S. Goudarzi, M. A. Asghari, S. Pichiah, G. S. Selopal, F. Rosei, Z. M. Wang and H. Zarrin, *ACS Applied Nano Materials*, 2021, **4**, 11323-11352.

89. Z.-w. Zhang, R.-t. Guo, J.-y. Tang, Y.-f. Miao, J.-w. Gu and W.-g. Pan, *Journal of CO2 Utilization*, 2021, **45**, 101453.

90. T. zhang, M. Maihemllti, K. Okitsu, D. Talifur, Y. Tursun and A. Abulizi, *Applied Surface Science*, 2021, **556**, 149828.

91.   D. Liu, D. Chen, N. Li, Q. Xu, H. Li, J. He and J. Lu, *Angewandte Chemie International Edition*, 2020, **59**, 4519-4524.

92.   Y. Xi, W. Mo, Z. Fan, L. Hu, W. Chen, Y. Zhang, P. Wang, S. Zhong, Y. Zhao and S. Bai, *Journal of Materials Chemistry A*, 2022, **10**, 20934-20945.

93.   J. Ma, L. Xu, Z. Yin, Z. Li, X. Dong, Z. Song, D. Chen, R. Hu, Q. Wang, J. Han, Z. Yang, J. Qiu and Y. Li, *Applied Catalysis B: Environment and Energy*, 2024, **344**, 123601.

94.   J. Wu, K. Li, S. Yang, C. Song and X. Guo, *Chemical Engineering Journal*, 2023, **452**, 139493.

95.   X. Wei, M. U. Akbar, A. Raza and G. Li, *Nanoscale Advances*, 2021, **3**, 3353-3372.

96.   Y. Lu, D. Wu, Y. Qin, Y. Xie, Y. Ling, H. Ye and Y. Zhang, *Separation and Purification Technology*, 2022, **302**, 122090.

97.   C. Yan, M. Xu, W. Cao, Q. Chen, X. Song, P. Huo, W. Zhou and H. Wang, *Journal of Environmental Chemical Engineering*, 2023, **11**, 111479.

98.   Z. Qian, L. Zhang, Y. Zhang and H. Cui, *Separation and Purification Technology*, 2023, **324**, 124581.

99.   D. Qin, Y. Zhou, W. Wang, C. Zhang, G. Zeng, D. Huang, L. Wang, H. Wang, Y. Yang, L. Lei, S. Chen and D. He, *Journal of Materials Chemistry A*, 2020, **8**, 19156-19195.

100. J.-y. Zhu, Y.-p. Li, X.-j. Wang, J. Zhao, Y.-s. Wu and F.-t. Li, *ACS Sustainable Chemistry & Engineering*, 2019, **7**, 14953-14961.

101. Z. Miao, Q. Wang, Y. Zhang, L. Meng and X. Wang, *Applied Catalysis B: Environmental*, 2022, **301**, 120802.

102. H. Y. Yin, Y. F. Zheng and X. C. Song, *RSC Advances*, 2019, **9**, 11005-11012.

103. Z. Zhang, L. Li, Y. Jiang and J. Xu, *Inorganic Chemistry*, 2022, **61**, 3351-3360.

104. W. Sun, J. Liu, F. Ran, N. Li, Z. Li, Y. Li and K. Wang, *Dalton Transactions*, 2024, **53**, 14018-14027.

105. Z. Wang, B. Cheng, L. Zhang, J. Yu and H. Tan, *Solar RRL*, 2022, **6**, 2100587.

106. A. H. Bhosale, S. Narra, S. S. Bhosale and E. W.-G. Diau, *The Journal of Physical Chemistry Letters*, 2022, **13**, 7987-7993.

107. N. Sun, M. Zhou, X. Ma, Z. Cheng, J. Wu, Y. Qi, Y. Sun, F. Zhou, Y. Shen and S. Lu, *Journal of CO2 Utilization*, 2022, **65**, 102220.

108. C. Yang, Q. Zhang, W. Wang, B. Cheng, J. Yu and S. Cao, *Science China Materials*, 2024, **67**, 1830-1838.

109. T. Li, R. Tao, Y. Wang, T. Yan, X. Fan and K. Liu, *Journal of Colloid and Interface Science*, 2024, **655**, 841-851.

110. Z. Wang, B. Cheng, L. Zhang, J. Yu, Y. Li, S. Wageh and A. A. Al-Ghamdi, *Chinese Journal of Catalysis*, 2022, **43**, 1657-1666.

111. H. Fu, T. Zhang, A. Abulizi, K. Okitsu and Y. Tursun, *Optical Materials*, 2023, **139**, 113790.

112. H. Li, D. Wang, C. Miao, F. xia, Y. Wang, Y. Wang, C. Liu and G. Che, *Journal of Environmental Chemical Engineering*, 2022, **10**, 108201.

113. X. Liu, S. Gu, Y. Zhao, G. Zhou and W. Li, *Journal of Materials Science & Technology*, 2020, **56**, 45-68.

114. J. H. Kim, G. Magesh, H. J. Kang, M. Banu, J. H. Kim, J. Lee and J. S. Lee, *Nano Energy*, 2015, **15**, 153-163.

115. Q. Shi, Z. Li, L. Chen, X. Zhang, W. Han, M. Xie, J. Yang and L. Jing, *Applied Catalysis B: Environmental*, 2019, **244**, 641-649.

116. R. Kavitha and S. G. Kumar, *Chemical Papers*, 2020, **74**, 717-756.

117. R. Kavitha and S. G. Kumar, *Materials Science in Semiconductor Processing*, 2019, **93**, 59-91.

118. C. Zhou, S. Wang, Z. Zhao, Z. Shi, S. Yan and Z. Zou, *Advanced Functional Materials*, 2018, **28**, 1801214.

119. Z. Duan, X. Zhao, C. Wei and L. Chen, *Applied Catalysis A: General*, 2020, **594**, 117459.

120. Z. Zhu, C.-X. Yang, Y.-T. Hwang, Y.-C. Lin and R.-J. Wu, *Materials Research Bulletin*, 2020, **130**, 110955.

121. Z.-H. Wei, Y.-F. Wang, Y.-Y. Li, L. Zhang, H.-C. Yao and Z.-J. Li, *Journal of CO2 Utilization*, 2018, **28**, 15-25.

122. X. Yue, L. Cheng, J. Fan and Q. Xiang, *Applied Catalysis B: Environmental*, 2022, **304**, 120979.

123. X. Wang, Y. Wang, M. Gao, J. Shen, X. Pu, Z. Zhang, H. Lin and X. Wang, *Applied Catalysis B: Environmental*, 2020, **270**, 118876.

124. M. Ahmadi, S. M. Alavi and A. Larimi, *Surfaces and Interfaces*, 2024, **45**, 103908.

125. L.-X. Liu, J. Fu, L.-P. Jiang, J.-R. Zhang, W. Zhu and Y. Lin, *ACS Applied Materials & Interfaces*, 2019, **11**, 26024-26031.

126. S. Yoshino, K. Sato, Y. Yamaguchi, A. Iwase and A. Kudo, *ACS Applied Energy Materials*, 2020, **3**, 10001-10007.

127. X. Y. Kong, W. Q. Lee, A. R. Mohamed and S.-P. Chai, *Chemical Engineering Journal*, 2019, **372**, 1183-1193.

128. S. Girish Kumar and K. S. R. Koteswara Rao, *Applied Surface Science*, 2015, **355**, 939-958.

129. J. F. de Brito, C. Genovese, F. Tavella, C. Ampelli, M. V. Boldrin Zanoni, G. Centi and S. Perathoner, *ChemSusChem*, 2019, **12**, 4274-4284.

130. K. Du, G. Li, J. Yang and S. Zhang, *Powder Technology*, 2024, **439**, 119656.

131. J. Liu, T. Du, P. Chen, Q. Yue, H. Wang, L. Zhou and Y. Wang, *Applied Surface Science*, 2024, **664**, 160274.

132. A. Rana, Sonu, A. Sudhaik, A. Chawla, P. Raizada, A. K. Kaushik, T. Ahamad, S. Kaya, N. Kumar and P. Singh, *Industrial & Engineering Chemistry Research*, 2024, **63**, 6960-6973.

133. Y. Feng, D. Chen, Y. Zhong, Z. He, S. Ma, H. Ding, W. Ao, X. Wu and M. Niu, *ACS Applied Materials & Interfaces*, 2023, **15**, 9221-9230.

134. Z. Li, Z. Li, J. Liang, W. Fan, Y. Li, Y. Shen, D. Huang, Z. Yu, S. Wang and Y. Hou, *Separation and Purification Technology*, 2023, **310**, 123197.

135. L. Zhao, H. Hou, L. Wang, C. R. Bowen, J. Wang, R. Yan, X. Zhan, H. Yang, M. Yang and W. Yang, *Chemical Engineering Journal*, 2024, **480**, 148033.

136. Y. Gong, L. Shen, Z. Kang, K. Liu, Q. Du, D. Ye, H. Zhao, X. A. Sun and J. Zhang, *Journal of Materials Chemistry A*, 2020, **8**, 21408-21433.

137. G. Li, Y. Li, H. Liu, Y. Guo, Y. Li and D. Zhu, *Chemical Communications*, 2010, **46**, 3256-3258.

138. C. Yang, Y. Wang, J. Yu and S. Cao, *ACS Applied Energy Materials*, 2021, **4**, 8734-8738.

139. J. Li, Y. Zhao, M. Xia, H. An, H. Bai, J. Wei, B. Yang and G. Yang, *Applied Catalysis B: Environmental*, 2020, **261**, 118244.

140. Y. Li, J. Jiang, C. Zhu, W. Feng, S. Han and L. Li, *Journal of Environmental Sciences*, 2025, **154**, 665-677.

141. J. Li, F. Wei, Z. Xiu and X. Han, *Chemical Engineering Journal*, 2022, **446**, 137129.

142. Q. Tang, W. Tao, J. Hu, T. Gui, Z. Wang, Y. Xiao, R. Song, Y. Jiang and S. Guo, *ACS Applied Nano Materials*, 2023, **6**, 17130-17139.

143. Y. Liu, R. Zou, Z. Chen, W. Tu, R. Xia, E. I. Iwuoha and X. Peng, *ACS Catalysis*, 2024, **14**, 138-147.

144. Y.-Y. Li, J.-S. Fan, R.-Q. Tan, H.-C. Yao, Y. Peng, Q.-C. Liu and Z.-J. Li, *ACS Applied Materials & Interfaces*, 2020, **12**, 54507-54516.

145. L. Wu, S. Zheng, H. Lin, S. Zhou, A. Mahmoud Idris, J. Wang, S. Li and Z. Li, *Journal of Colloid and Interface Science*, 2023, **629**, 233-242.

146. M. Zhang, Y. Mao, X. Bao, G. Zhai, D. Xiao, D. Liu, P. Wang, H. Cheng, Y. Liu, Z. Zheng, Y. Dai, Y. Fan, Z. Wang and B. Huang, *Angewandte Chemie International Edition*, 2023, **62**, e202302919.

147. I. T. Chen, M.-W. Zheng, Y.-C. Pu and S.-H. Liu, *International Journal of Energy Research*, 2022, **46**, 15198-15210.

148. J. Su, G.-D. Li, X.-H. Li and J.-S. Chen, *Advanced Science*, 2019, **6**, 1801702.

149. H. Wei, F. Meng, H. Zhang, W. Yu, J. Li and S. Yao, *Journal of Materials Science & Technology*, 2024, **185**, 107-120.

150. J. He, J. Lin, Y. Zhang, Y. Hu, Q. Huang, G. Zhou, W. Li, J. Hu, N. Hu and Z. Yang, *Chemical Engineering Journal*, 2024, **480**, 148036.

151. W. Dai, W. Xiong, J. Yu, S. Zhang, B. Li, L. Yang, T. Wang, X. Luo, J. Zou and S. Luo, *ACS Applied Materials & Interfaces*, 2020, **12**, 25861-25874.

152. B. Yu, Y. Wu, F. Meng, Q. Wang, X. Jia, M. Wasim Khan, C. Huang, S. Zhang, L. Yang and H. Wu, *Chemical Engineering Journal*, 2022, **429**, 132456.

153. Y. Liu, X. Kong, X. Guo, Q. Li, J. Ke, R. Wang, Q. Li, Z. Geng and J. Zeng, *ACS Catalysis*, 2020, **10**, 1077-1085.

154. M. Ahmadi, S. M. Alavi and A. Larimi, *Inorganic Chemistry*, 2023, **62**, 20372-20389.

155. Y. Zhou, W. Jiao, Y. Xie, F. He, Y. Ling, Q. Yang, J. Zhao, H. Ye and Y. Hou, *Journal of Colloid and Interface Science*, 2022, **608**, 2213-2223.

156. S. Li, N. Hasan, F. Zhang, J.-S. Bae and C. Liu, *Journal of Colloid and Interface Science*, 2023, **652**, 1533-1544.

157. Y. Zhang, S. Liu, X. Guo, H. Mikulčić, R. Xiao and X. Wang, *Journal of Photochemistry and Photobiology A: Chemistry*, 2024, **447**, 115278.

158. Y. Zhu, R. Wu, J. Hui, Z. Zhang and S. Wei, *Journal of Alloys and Compounds*, 2024, **1003**, 175609.

159. X. Bai, L. He, W. Zhang, F. Lv, Y. Zheng, X. Kong, D. Wang and Y. Zhao, *Journal*, 2023, **13**.

160. M. Guo, Z. Wang, X. Wu, J. Qiu, L. Gu and Z. Yang, *ACS Applied Energy Materials*, 2023, **6**, 2863-2876.

161. Y. Wu, J. Xiao, J. Yuan, L. Wang, S. Luo, Z. Zhang, X. Fu and W. Dai, *Journal of Colloid and Interface Science*, 2024, **674**, 158-167.

162. Y. Xie, C. Qiu, L. Wang, Y. Wang, J. Zhang, J. Zhang, H. Wan and G. Guan, *Ceramics International*, 2024, **50**, 25161-25169.

163. C. Zhou, X. Shi, D. Li, Q. Song, Y. Zhou, D. Jiang and W. Shi, *Journal of Colloid and Interface Science*, 2021, **599**, 245-254.

164. X. Hu, J. Hu, Q. Peng, X. Ma, S. Dong and H. Wang, *Materials Research Bulletin*, 2020, **122**, 110682.

165. S. Song, Z. Song, H. Han, K. Wei, W. Zhang, D. Liu, Q. Wang, C. Ma, S. Feng and X. Duan, *Applied Catalysis A: General*, 2024, **683**, 119834.

166. W. Wang, X. Feng, L. Chen and F. Zhang, *Industrial & Engineering Chemistry Research*, 2021, **60**, 18384-18396.

167. M. Sun, K. Fan, C. Liu, T. Gui, C. Dai, Y. Jia, X. Liu and C. Zeng, *Langmuir*, 2024, **40**, 12681-12688.

168. X. Yu, F. Wen, F. Zhang, P. Yang, Y. Zhao, Y. Wu, Y. Wang and Z. Liu, *ChemSusChem*, 2020, **13**, 5565-5570.

169. Y. Sun, X. Qi, R. Li, Y. Xie, Q. Tang and B. Ren, *Optical Materials*, 2020, **108**, 110170.

170. H. Wang, H. Zhu, Y. Nie, X. Zhang and G. Xiang, *ACS Applied Materials & Interfaces*, 2025, **17**, 39218-39225.

171. R. Fang, Z. Yang, Z. Wang, J. He, J. Qiu, J. Ran and R. Godin, *Fuel*, 2024, **367**, 131514.

172. S. G. Kumar and K. S. R. K. Rao, *Nanoscale*, 2014, **6**, 11574-11632.

173. M. Sahu and C. Park, *Materials Today Sustainability*, 2023, **23**, 100441.

174. X. Zhao, Z. Duan and L. Chen, *Industrial & Engineering Chemistry Research*, 2019, **58**, 10402-10409.

175. X. Jia, J. Cao, H. Sun, X. Li, H. Lin and S. Chen, *Applied Catalysis B: Environmental*, 2024, **343**, 123522.

176. K. Wang, X. Feng, Y. Shangguan, X. Wu and H. Chen, *Chinese Journal of Catalysis*, 2022, **43**, 246-254.

177. S. She, B. Zhao, J. Wang, Z. Wei, X. Wu and Y. Li, *Separation and Purification Technology*, 2023, **309**, 123064.

178. H. Tang, X. Wang, C. Yao and Z. Zhang, *Separation and Purification Technology*, 2024, **338**, 126544.

179. Y. Huo, J. Zhang, Z. Wang, K. Dai, C. Pan and C. Liang, *Journal of Colloid and Interface Science*, 2021, **585**, 684-693.

180. D.-E. Lee, N. Mameda, K. P. Reddy, B. M. Abraham, W.-K. Jo and S. Tonda, *Journal of Materials Science & Technology*, 2023, **161**, 74-87.

181. P. J. Mafa, M. E. Malefane, A. O. Oladipo, S. L. Lebelo, D. Liu, J. Gui, B. B. Mamba and A. T. Kuvarega, *Journal of Environmental Chemical Engineering*, 2025, **13**, 117821.

182. S. Zhang, Y. Si, B. Li, L. Yang, W. Dai and S. Luo, *Small*, 2021, **17**, 2004980.

Data Availability

No data used in this article.

RSC Applied Interfaces Accepted Manuscript