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# HF-free microwave-assisted synthesis of MXene as an electrocatalyst for hydrogen evolution in alkaline media†

MXenes, characterized by their robustness, flexibility, and large surface-to-volume ratio facilitating efficient energy transfer with fast response times, have emerged as promising electrocatalysts for hydrogen generation through electrochemical water-splitting. However, the conventional synthetic route to MXenes typically involves the use of hydrofluoric acid (HF) to obtain MXenes with terminal F-functional groups. Unfortunately, these fluorine groups can negatively impact the electrocatalytic performance of MXenes. Moreover, HF is highly toxic, necessitating the development of more environmentally friendly synthetic methods. In response to these challenges, we have developed a novel HF-free microwaveassisted synthesis approach for MXenes. This method harnesses the benefits of uniform heating, homogeneous nucleation, and rapid crystal development, resulting in MXene crystallites with limited size. Importantly, our microwave-assisted approach utilizes a fluoride-free, less hazardous etchant as compared to HF for the synthesis and functionalization of MXene. The as-obtained MXene exhibits significantly improved performance towards the electrochemical hydrogen evolution reaction in alkaline media. Specifically, it demonstrates an overpotential of 140 mV at a current density of 10 mA cm<sup>-2</sup> and a Tafel slope of 84 mV dec<sup>-1</sup>. These results highlight the potential of our HF-free microwave-assisted synthesis approach for producing high-quality MXenes with enhanced electrocatalytic activity for hydrogen generation.

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# Introduction

Conventional energy sources have been crucial in meeting the world's energy demands, but their negative impact on the environment and human health necessitates a shift towards alternative, sustainable options such as solar, wind, biomass, and water energy.¹ Electrochemical water splitting emerges as a promising technique for producing high-purity hydrogen without carbon dioxide emissions, offering a vital solution to contemporary environmental challenges.² However, the most efficient catalysts for the hydrogen evolution reaction (HER), primarily noble metal-based electrocatalysts such as platinum (Pt) and ruthenium (Ru), are hindered by their high cost and scarcity.⁴ Indeed, researchers are increasingly focusing on the development of low-cost catalysts with high electro-catalytic

activity to address the growing demand for sustainable energy

In recent years, MXenes, among various 2D materials, have garnered significant attention as catalysts for enhancing electrocatalytic hydrogen production. MXene's excellent conductivity, hydrophilic surface, high surface area, tuneable terminal

solutions and to optimize the catalytic performance. 10-13 Among the various materials being explored, two-dimensional (2D) materials have garnered significant interest due to their unique physicochemical properties, atomic-scale thickness, and ease of structural modification.14-18 The atomic-scale thickness of 2D materials offers several advantages, including a high surface-tovolume ratio, which exposes a larger number of active sites for catalytic reactions. 19-21 Additionally, their tuneable electronic and chemical properties make them highly versatile for various applications. Moreover, the ease of structural modification allows researchers to tailor the properties of 2D materials to suit specific catalytic requirements, further enhancing their performance.<sup>22-24</sup> Overall, 2D materials hold immense potential for catalysis due to their unique characteristics, ease of reaction condition optimization, and exciting electrical properties making them promising candidates for the development of lowcost and efficient catalysts for a wide range of electrochemical processes, including the hydrogen evolution reaction and oxygen reduction reaction, among others. 13,25,26

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groups, and numerous reaction sites make it an attractive candidate for the HER. Furthermore, the prolonged lifetime of charge carriers and simple morphological configuration profoundly enhance the catalytic activity.27,28 The interest in MXenes, particularly Ti-based MXenes, has grown rapidly since the discovery of Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> in 2011.<sup>29</sup> This is mainly attributed to their ease of synthesis and exceptional properties. However, research on MXenes is not limited to Ti systems, and exploration for MXenes beyond titanium is currently underway. 24,25,30 Typically, MXene synthesis involves selectively etching the "A" element from the MAX phase, 31,32 where M is an early transition metal, A is a group IIIA or IVA element, X is C and/or N, and n =1, 2, or 3.33 Traditional methods of MXene synthesis, such as using hydrofluoric acid (HF) or molten salts, pose challenges due to safety concerns and high energy consumption.34-36 Efforts are being made to develop alternative synthesis routes for MXenes that address these challenges and offer safer and more energy-efficient processes. These advancements are crucial for furthering the utilization of MXenes in various applications, including electrocatalytic hydrogen production and beyond.

In this work, we introduce a novel microwave-assisted method for synthesizing MXene, bypassing the need for hazardous HF etchants. Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>, as MXene, has outstanding stability and electrochemical performance. However, its enormous potential has never been completely realised due to restrictions in the self-weight stacking of the layered structure, resulting in inadequate ion-accessible surface area for the electrolyte. MXene production and exfoliation using traditional techniques typically produce thicker MXene sheets with lower lateral dimensions. However, in this work all of the aforementioned constraints can be addressed by employing the microwave-assisted approach. This study presents a novel approach for obtaining a larger lateral size few-layer MXene in a much shorter period (just 30 minutes) than typical etching procedures, by employing microwave radiation to quickly evaporate interlayer water.37 The microwave-assisted technique offers several advantages, including rapid crystallization, costeffectiveness, ease of operation, phase selectivity, controlled material size, rapid heating, high reaction rates, and morphology control.38,39 Our unique microwave-assisted fluoride-free NaOH-based chemical reduction technique enables quick reactions and intercalates the ions into the sheets, resulting in the formation of crumpled sheet-like morphology MXene. The key improvements in the MXene synthesized with microwave assistance include elimination of the need for a hydrothermal reactor, allowing for the reaction to take place in an open vessel. Lastly, utilization of an alkali (NaOH) as an innocuous etchant compared to hazardous HF based etching and intercalated agents ensures safety and quite environmental friendliness. The resultant microwave-assisted MXene, referred to as MMXene (where "M" denotes microwave), has been characterized using various physical analysis techniques, confirming the successful synthesis of MXene with crumpled sheet-like morphology. When evaluated for electrochemical water splitting, 2D MMXene nanosheets exhibited exceptional performance in hydrogen evolution reaction (HER)

activity, demonstrating a low overpotential of 140 mV at a current density of 10 mA cm<sup>-2</sup> and a Tafel slope of 84 mV dec<sup>-1</sup>. To our knowledge, this is the first report of implementing a non-HF-based microwave-assisted synthesis of MXene nanosheets for electrocatalytic hydrogen evolution. These findings highlight the potential of our approach to produce high-quality MXene catalysts for various applications in sustainable energy conversion.

# 2. Experimental section

#### 2.1 Materials and methods

MAX phase titanium aluminium carbide  $(Ti_3AlC_2)$  was purchased from Nanoshell Co. Ltd, India, while lithium fluoride (LiF), 98.5%, was obtained from Alfa Aesar. Nafion solution was purchased from the fuel cell store, and nickel foam was sourced from IPGI Chennai, India. All other chemicals of analytical grade were acquired from Finar Chemicals, India, and were used without additional purification. Deionized water was utilized in all experiments to ensure consistency and accuracy of results.

2.1.1 Microwave-assisted synthesis of MXene. MXene  $(Ti_3C_2)$  was synthesized via a microwave-assisted reduction method as illustrated in Fig. 1. In this process, MAX phase titanium aluminium carbide  $(Ti_3AlC_2)$  was utilized as the precursor for the formation of MXene  $Ti_3C_2$ , while sodium hydroxide (NaOH) served as both an etchant and a reducing agent. In brief, 100 ml of 2 M NaOH solution and 1 g of  $Ti_3AlC_2$  precursor were mixed and placed in a microwave at 210 W for 30 minutes. Following the microwave treatment, the resulting black powder was collected and subjected to repeated washing with ethanol and distilled water. Finally, the washed products were dried at 60 °C in an oven overnight, resulting in the formation of blackish powder of MMXene.

**2.1.2 LiF synthesis approach for MXene.** Additionally,  $\mathrm{Ti_3C_2T_x}$  was synthesized via a modified etching route following earlier reports. In this method, 1 g of LiF was dissolved in 20 ml of HCl in a 100 ml polypropylene plastic vial to create the etchant solution for the LiF technique. Subsequently, 1 g of  $\mathrm{Ti_3AlC_2}$  precursor was slowly added to the mixture, and the reaction was allowed to proceed for 24 hours at 35 °C. The acidic product obtained was thoroughly washed with deionized water (DI H<sub>2</sub>O) via centrifugation at 3500 rpm until the pH of the solution reached or exceeded 6. Following centrifugation for 1 hour, a dark black solution containing large  $\mathrm{Ti_3C_2T_x}$  flakes was collected. Lie and  $\mathrm{Ti_3C_2T_x}$  flakes was collected.

#### 2.2 Materials characterization

The scanning electron microscopy (FE-SEM) analysis was conducted to examine the morphology of the materials. The FE-SEM utilized was model JSM-IT200 by JEOL Ltd, operated at a voltage of 20 kV. For examining the crystalline structures of the materials, X-ray diffraction (XRD) analysis was performed using a D2 Phaser model with a diffractometer using Cu target ( $\lambda = 1.54 \,\text{Å}$ ) radiation, with a scanning range from 3 to 160° and

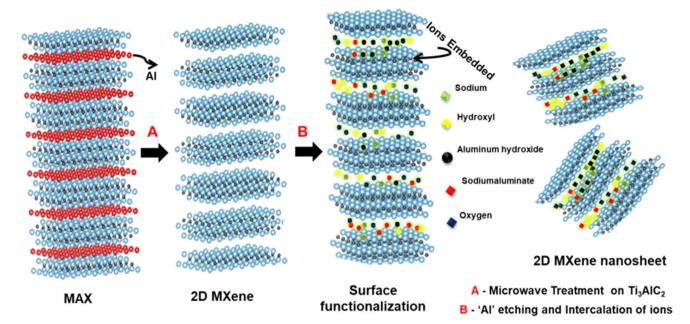


Fig. 1 Schematic representation of microwave assisted synthesis of MXene.

a step scan of  $0.05^{\circ}$ . Elemental analysis was carried out using FTIR PerkinElmer analytical instruments.

#### 2.3 Electrochemical measurements

First, nickel foam (NF) (length  $\times$  breadth = 1  $\times$  1 cm, IPGI instruments, Chennai, India) was cleaned by sonication using hydrochloric acid, ethanol and water for 15 min each and dried in an oven. 10 mg of MMXene was synthesised by using the microwave-assisted approach, and 10 µl of Nafion solution and  $400 \mu l$  of water were mixed and sonicated to obtain a uniform slurry. We employed Ni foam as a substrate; however, distinct from other substrates, Ni foam is porous, has a high catalytically active surface area, and is resistant to corrosion in alkaline solution. 42 This slurry (200 μl) was then cast drop by drop on Ni foam (Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>/NF), which was then dried on a hot plate to obtain 20 mg mass deposition and finally dried at 60 °C in an oven for 1 h. Hydrogen evolution activity of the catalysts was measured using an electrochemical workstation (Autolab Instrument Inc.) in a conventional three-electrode configuration with 2 M NaOH electrolyte (99.99%) at room temperature. Glassy carbon is used as a counter electrode material in threeelectrode setups for electroanalytical chemistry. A silver chloride electrode (Ag/AgCl) was used as the reference electrode and Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>/NF was used as the working electrode. Before electrochemical measurements, 2 M NaOH was purged with highly pure N2 gas (99.99%) to remove any interfering gases such as  $H_2$ ,  $O_2$ , etc. All data were presented after the conversion to the reversible hydrogen electrode (RHE) scale using the Nernst equation ( $E = E_0 + E_{Ag/AgCl} + 0.059 \times pH, E_{Ag/AgCl} = 0.1976 V$ ). The electrocatalytic HER performance of the samples was evaluated by linear sweep voltammetry (LSV) and cyclic voltammetry (CV) with a scanning rate of 500 mV s<sup>-1</sup> with potential range -1.4 to 0 V and -2 to 1.5 V respectively. The

electrochemical impedance spectroscopy (EIS) was conducted (Nyquist plot) in a frequency range of 1000 kHz to 0.01 Hz and at a voltage of 1  $\rm V$ .

# Results and discussion

#### 3.1 Physical characterization

The XRD analysis provided valuable insights into the structural changes occurring during the synthesis of MXene from the MAX phase Ti<sub>3</sub>AlC<sub>2</sub>. The characteristic peaks observed in the XRD patterns confirm the presence of specific crystallographic planes in both MAX and MXene phases as shown in Fig. 2a, where the characteristic peaks at 9.8° and 38.9° correspond to (002) of stacked layers in the MAX phase and (104) of 'Al' atomic arrangement, respectively.43 The shift in the (002) peak towards a lower angle in the MXene compared to the MAX phase indicates an increase in the interlayer spacing, suggesting successful extraction of aluminium atoms and formation of MXene. The near disappearance of the (104) peak further supports the removal of aluminium from the Ti<sub>3</sub>AlC<sub>2</sub> MAX phase.<sup>44</sup> The narrow peaks in the XRD pattern indicate the crystalline nature of the MMXene sheets, with well-arranged atoms or ions in highly ordered structures. Overall, the XRD data provide compelling evidence for the successful synthesis of MXene from the Ti<sub>3</sub>AlC<sub>2</sub> MAX phase.

The FTIR spectrum of MMXene revealed several bands corresponding to different functional groups as shown in Fig. 2b. These include Ti–C, Ti–O, C–Cl, Al–OH, C=C, C=O, O=C=O, C–H, and Na–OH at wavenumbers 525, 682, 850, 1060, 1610, 1820, 2111, 2340, 2668, 3130 cm<sup>-1</sup>, respectively. The presence of these bands indicates the presence of various chemical bonds and functional groups in the MMXene structure. For instance, the Ti–C and Ti–O bands suggest the presence of titanium–

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a) (MAX) (No. 1) (MAX) (No. 1) (MAX) (MAX)

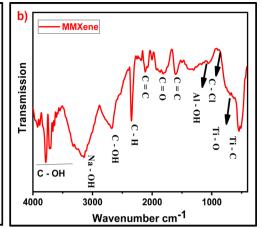


Fig. 2 Physical characterization of MMXene, (a) XRD and (b) FTIR.

carbon and titanium-oxygen bonds, respectively. The C-Cl indicates the presence of chlorine-containing compounds, while the presence of C=C and C=O bands indicates the presence of carbon-carbon double bonds and carbonyl groups, respectively. The O=C=O band suggests the presence of carbon dioxide due to an open environment reaction. The C-H band indicates the presence of carbon-hydrogen bonds, and the Na-OH band suggests the presence of sodium hydroxide residues. Furthermore, the presence of C-OH bands in the region of 3500 to 3800 cm<sup>-1</sup> suggests the involvement of hydroxyl groups resulting from the reaction between the MAX phase and the solvent.45 The FTIR data also indicate the hydrophilic nature of MMXene due to the presence of oxygen and hydroxyl groups.46 However, the weak bands from these moieties suggest their low concentration or intensity. The presence of sodium, as indicated by the Na-OH band, further confirms the involvement of NaOH solution in the synthesis process. The EDX analysis provides complementary information regarding the elemental composition of MMXene, corroborating the findings from the FTIR analysis and confirming the high purity of the synthesized MXene.47

The successful synthesis of Ti<sub>3</sub>C<sub>2</sub> and the formation of a crumbled lamellar-like sheet structure were observed through Scanning Electron Microscope (SEM) analysis which is helpful for effective hydrogen generation application, as depicted in Fig. 3c. The basal planes of Ti<sub>3</sub>C<sub>2</sub> exhibited buffing due to microwave irradiation, attributed to the replacement of the Al atom layer by  $T_x$  (intercalated ions). The SEM image of MMXene clearly demonstrates the engineered crumbled lamellar-like sheet structure with desirable inter-planar spacing achieved through aluminium etching. In MMXene, the presence of excess O and Na signals can be attributed to surface terminations (-Na and/or -O and -OH) as well as intercalated water and sodium ions between Ti<sub>3</sub>C<sub>2</sub> layers, as revealed by Energy Dispersive Spectroscopy (EDS) analysis shown in Fig. 3d.48 The residual Al signal may originate from the reaction product Al(OH)<sub>4</sub>, which might not have been completely washed away. However, the EDS spectrum confirms that the chemical compositions of MMXene primarily consist of Ti, C, O, and Na.49

#### 3.2 Synthesis mechanism of MXene

In this study, a microwave-assisted route was employed for the synthesis of MXene, utilizing a HF-free etchant, sodium hydroxide, and MAX phase Ti<sub>3</sub>AlC<sub>2</sub> as the MXene precursor. This approach leverages the alignment of dipoles within the material induced by external fields generated by microwave electromagnetic radiation.50 When dielectric materials absorb microwaves, they undergo molecular motion, causing ionic species to migrate and dipolar species to rotate. Unlike heat convection and conduction, microwave heating promotes the fragmentation of large precursor molecules and accelerates the breakage of chemical bonds.<sup>51</sup> Importantly, microwave irradiation facilitates interactions among a higher number of ions, thereby enhancing catalytic processes in a shorter duration. This is considered a key advantage of microwave etching for obtaining desirable MXene materials. When MAX phase Ti<sub>3</sub>AlC<sub>2</sub> is introduced into a NaOH solution, it disperses in water. Irradiation induces friction with water, generating heat and leading to the dissociation of ions such as sodium (Na+) and hydroxyl  $(OH^{-})$ , as depicted in eqn (1).

$$NaOH + H_2O \rightarrow Na^+ + OH^- + H_2O$$
 (1)

Subsequently,  $Ti_3AlC_2$  undergoes dissociation and friction, leading to the breakage of Ti and Al bonds and the formation of  $Al_2O_3$ . This  $Al_2O_3$  then etches out from the parent MAX phase as sodium aluminate hydrate ions, as illustrated in eqn (2). The resulting sodium aluminate hydrate (NaAl(OH)<sub>4</sub>) can be readily separated due to its dissociation in water, resulting in the formation of  $Ti_3C_2T_x$ .

$$Al_2O_3 + 2NaOH + 3H_2O \rightarrow Ti_3C_2T_x + 2NaAl(OH)_4$$
 (2)

Accordingly, the remaining NaAl(OH)<sub>4</sub> further dissociates into Na<sup>+</sup> and Al(OH)<sub>4</sub><sup>-</sup>. As Al is etched out, these ions subsequently get intercalated into the MXene as hydroxide and sodium, which originate from the solution, along with oxygen from the air. As a result, we have successfully synthesized

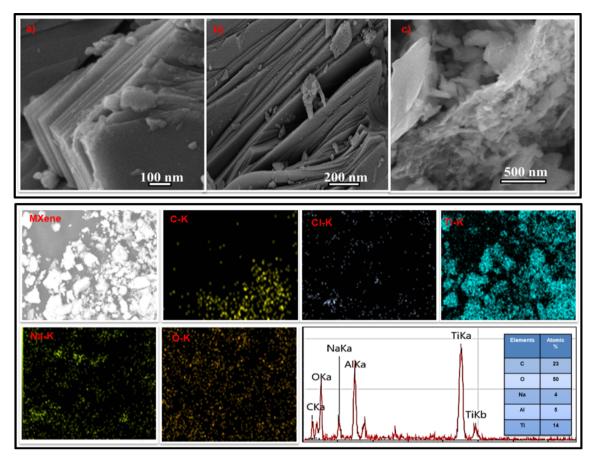


Fig. 3 SEM image of (a) MAX phase, (b) LiF MXene and (c) MMXene. (d) EDS and elemental mapping of MMXene.

MXene *via* a microwave-assisted route using HF-free etchant NaOH and Ti<sub>3</sub>AlC<sub>2</sub> (Table 1).

$$Ti_3C_2T_x + NaOH + H_2O \rightarrow Ti_3C_2T_x(T_x-Na/O/OH)$$
 (3)

#### 3.3 Electrochemical activity

To evaluate the electrocatalytic activity towards the hydrogen evolution reaction (HER), MMXene was drop-cast onto nickel foam (NF) using Nafion as a binder (mass loading: 20 mg cm<sup>-2</sup>). The MMXene@NF electrode materials were then analysed in a three-electrode electrochemical cell in 2 M NaOH through cyclic voltammetry (CV), linear sweep voltammetry (LSV), Tafel plot, and electrochemical impedance spectroscopy measurements. In the LSV in Fig. 4a, it is depicted that the hydrogen

evolution overpotential of MXene synthesized using LiF and MMXene as working electrodes at 10 mA cm $^{-2}$  is 370 and 140 mV, respectively. The lower overpotential of MMXene synthesized by the microwave-assisted approach could be attributed to the insertion of a number of sodium, oxygen, hydroxyl ions, and hydroxo aluminate ions, which participate in the catalytic reaction to accelerate the process. Furthermore, MMXene showed the highest current density in a potential window of -1.0 to 0 V. MMXene basal planes have numerous functionalized active sites, and LiF-assisted etching techniques promote fluorine functionalization on the basal planes, reducing the materials HER catalytic activity. Although hydrogen adsorption (GH) energy close to zero is preferred for efficient HER activity, density functional theory (DFT) studies of fluorine-functionalized MXene reveal that the GH value

Table 1 Comparison of microwave-assisted synthesised MXene to prior methods for electrochemical hydrogen generation in alkaline electrolytes

MXene	Synthesis approach	Electrolyte	Overpotential (mV)@10 mA cm <sup>-2</sup>	Tafel slop (mV dec <sup>-1</sup> )	References
$\mathrm{Ti}_3\mathrm{C}_2\mathrm{T}_x$	Microwave	NaOH	140	84	This work
	In situ HF	КОН	218	156	52
	HF	NaOH	1000	_	53
	In situ HF	КОН	344	216	54
	In situ HF	КОН	532	101	55
	In situ HF	КОН	541	468	56

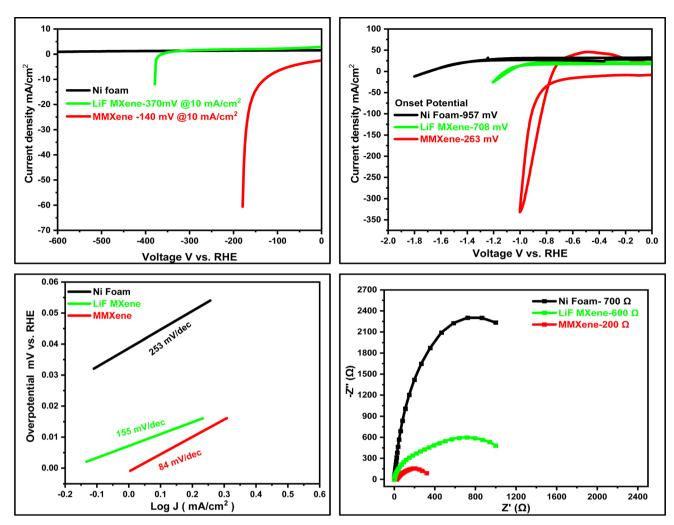


Fig. 4 Electrochemical characterization of MMXene, (a) LSV, (b) CV, (c) Tafel plot and (d) impedance spectra

increases with fluorine surface coverage.<sup>57</sup> Fluorine-free methods mitigate the adverse effects of –F terminations on MXene conductivity, improving electronic properties.<sup>58</sup> Furthermore, oxygen and hydroxyl functional groups have been shown to increase HER activity while significantly reducing overpotential.<sup>59</sup> As a result, fluorine-free procedures are strongly recommended for electrochemical hydrogen production.

CV was used to further investigate how the ions alter the kinetics of the hydrogen evolution reaction, which was recorded from -2 to 1.5 V at 100 mV s<sup>-1</sup>. As shown in Fig. 4b, a CV curve at 100 mV s<sup>-1</sup> for the bare Ni foam catalyst, LiF MXene (20 mg cm<sup>-2</sup> mass loading) and MMXene deposited on Ni foam (20 mg cm<sup>-2</sup> mass loading) was recorded. For the MMXene catalyst, the HER onset potential occurs at about 263 mV  $\nu$ s. RHE as indicated by a sudden increase in the cathodic current as shown in Fig. 4b, while for bare Ni foam and LiF MXene it occurs at 957 mV and 708 mV, respectively. The MMXene catalyst displayed better activity for hydrogen evolution with current density 62 mA cm<sup>-2</sup>, which is much higher as compared to that of bare Ni foam and LiF MXene. The better electrocatalytic activity of the MMXene electrode is mainly attributed to the thin

lamellar MMXene nanosheet with desirable distance between the Ti<sub>3</sub>C<sub>2</sub>T<sub>r</sub> layers and efficient electron transfer, resulting in better access of electrolyte to the catalytically active surface.<sup>61</sup> The surface area of the electrocatalyst was assessed using the capacitance approach, which involved analysing cyclic voltammetry (CV) data obtained at different scan rates, ranging from 50 to 200 mV s<sup>-1</sup>. The electrochemical surface area (ECSA) was subsequently calculated based on these measurements. This process entailed determining the double-layer capacitance (Cdl) values from the CV curves. The double-layer capacitance (Cdl) values for the samples were determined by graphing  $(|J_a| - |J_c|)$ against the cyclic voltammetry (CV) scan rate. This involved plotting the absolute difference between the anode current density  $(J_a)$  and the cathode current density  $(J_c)$  against the scan rate. The resulting graph allowed for the calculation of Cdl values, which are crucial for assessing the electrochemical surface area (ECSA) of the samples. 62 Furthermore, the observed improvements in electrochemical activity can be attributed to the composition of the MMXene, which lacks fluorine groups (F) and primarily consists of hydroxyl groups (-OH), sodium groups (-Na), oxygen groups (-O), and hydroxo aluminate

groups (–AlOH). To better understand the kinetics of the hydrogen evolution reaction (HER) catalysed by MMXene, the rate-determining steps were investigated using Tafel slope analysis, as shown in Fig. 4c. The Tafel slopes obtained for bare Ni foam, LiF MXene, and MMXene are 253, 155, and 84 mV dec<sup>-1</sup>, respectively. Considering that Tafel analysis was conducted using the current response at low overpotential, it is likely that the Volmer step serves as the rate-determining step. This change in the reaction mechanism on MMXene suggests that intercalated ions play a crucial role in accessing the surface for hydrogen adsorption and desorption. It is widely accepted that the HER under alkaline conditions proceeds *via* a three-step mechanism.<sup>63,64</sup>

$$M(Ti_3C_2T_x) + H_2O + e^- = MH_{ads} + OH^-$$
, Volmer reaction (4)

$$MH_{ads} + H_2O + e^- = H_2 + M + OH^-$$
, Heyrovsky reaction (5)

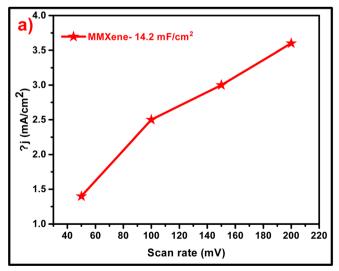
$$MH_{ads} + MH_{ads} = H_2 + M$$
, Tafel reaction (6)

As illustrated in Fig. 4d, electrochemical impedance spectroscopy (EIS) was employed to investigate the charge carrier recombination/transfer behaviour of MMXene. The MMXene electrocatalyst demonstrated the smallest arc size on the Nyquist curve, indicating high conductivity that facilitates easy electron migration. In contrast, LiF MXene exhibited a larger radius of the arc, suggesting an increase in carrier transfer impedance. In the Nyquist diagram, the impedance value reflects the resistance in the transfer of charges during the hydrogen evolution reaction (HER). A smaller arc radius corresponds to a lower impedance value of the test sample, indicating a greater likelihood of the hydrogen evolution catalytic reaction occurring. The charge transfer resistance is determined by the size of the semicircle and the properties of the electrode surface. The charge transfer resistance values for pure Ni foam, LiF MXene, and MMXene are 700, 600, and 200  $\Omega$ ,

respectively. This suggests that MMXene exhibits significantly lower charge transfer resistance compared to Ni foam and LiF MXene, further highlighting its superior electrochemical performance in facilitating the hydrogen evolution reaction.

To account for the challenge in determining the mass of the etched layer, the electrochemical surface area (ECSA) values were represented per unit of surface area, expressed as real capacitances (F cm<sup>-2</sup>). This approach allows for a standardized comparison of ECSA values across different samples, irrespective of variations in the mass of the etched layer. As depicted in Fig. 5a, the measured double-layer capacitance (Cdl) for the MMXene electrocatalyst was found to be 14.2 mF cm<sup>-2</sup>. This increased average Cdl value compared to earlier reports suggests that the MMXene offers a larger electro-active surface area, which is beneficial for electrochemical performances. 65 Additionally, Fig. 5b demonstrates the excellent stability of MMXene in alkaline electrolyte. The potential of the catalyst remains stable at 260 mV during continuous electrolysis at 10 mA cm<sup>-2</sup> for a duration of 9 hours. This stability underscores the potential utility of MMXene as an electrocatalyst in various applications. The voltage gradually decreased from 315 to 266 mV to maintain a current density of 10 mA cm<sup>-2</sup> for the first hour and 15 min. There was no significant drop in potential for the remaining 9 hours and 25 min. Over a period of time, the potential remains constant for up to almost 9 hours and 25 min, indicating that the synthesized catalyst is stable in electrochemical hydrogen production with no potential reduction.

To analyse the structural stability of the MMXene/Ni electrode, we conducted XRD and FE-SEM analysis after a long term stability test as shown in Fig. S1.† Fig. S1a† shows the XRD pattern of the MMXene/Ni electrode before and after the electrochemical stability test (after 10 h). There were minor differences observed, such as the characteristic diffraction peak of the MMXene nanostructure (indicated by \*) becoming broadened due to long-term ion intercalation into the layered sites of the MMXene nanostructures. Fig. S1b† confirms that there is no



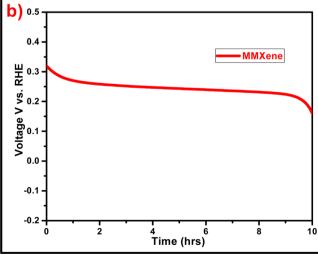


Fig. 5 (a) ECSA plots inferred from CV curves at different scan rates. (b) Chronopotentiometry curve at a constant current density of  $10 \text{ mA cm}^{-2}$  for the MMXene catalyst in 2 M NaOH.

significant change in the morphology of the MMXene/Ni electrode after electrochemical stability tests, which highlights the good adherence of the MMXene on the Ni foam. The ion adsorption on the surface of MXene converts the morphology of the thin crumbled layer to a thick, rough surface. The XRD and FE-SEM results indicate that the MMXene/Ni foam possesses good structural stability in an alkaline electrolyte towards the hydrogen evolution reaction.

# 4. Conclusion

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In summary, MXene with 2D structures was prepared using the microwave-assisted etching method, and its electrochemical activities towards hydrogen evolution were investigated. This novel microwave-assisted approach offers increased reaction kinetics, rapid initial heating, and enhanced reaction rates, resulting in clean reaction products. Notably, this approach required less time (30 minutes) for the reaction, with the functionalization of MMXene facilitated by the presence of several functional groups. The as-synthesized MMXene exhibited outstanding catalytic activity in alkali electrolyte, as demonstrated by EDS analysis revealing the presence of various terminations in MMXene, which aids in more efficient electron transfer. In contrast to earlier synthesis approaches for MXene involving HF or in situ HF etching, microwave-assisted synthesized MXene overcomes drawbacks such as the use of toxic chemicals, fluorine functionalization, amorphous nature, unstable structure, and inadequate electron transport. Microwave-assisted synthesized MXene has been identified as an efficient electrocatalyst for hydrogen generation through electrochemical water splitting compared to previously reported MXenes. The excellent performance of the  $Ti_3C_2T_x$  (O, OH AlOH, and Na ions) catalyst in the hydrogen evolution reaction (HER) can be attributed to several key characteristics: high coverage of catalytically active O terminations on the basal planes, ultrathin nanosheet architecture exposing more active sites, and the presence of numerous terminations. Furthermore, F-free MXene from this study exhibits effective electronic conductivity, facilitating excellent charge transfer, which is crucial for efficient hydrogen generation.

#### Conflicts of interest

All data underlying the results are available as part of the article itself and no additional date are required.

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