

Cite this: *Chem. Sci.*, 2022, 13, 12367 All publication charges for this article have been paid for by the Royal Society of Chemistry

# Precisely synthesized LiF-tipped CoF<sub>2</sub>-nanorod heterostructures improve energy storage capacities†

Siyuan Wang,<sup>a</sup> Hao Fu,<sup>ab</sup> Jiamin Ma,<sup>a</sup> Xiaomeng Shi,<sup>a</sup> Huimin Wang,<sup>c</sup> Zongyou Yin,<sup>d</sup> Shuai Zhang,<sup>a</sup> Mengdie Jin,<sup>a</sup> Ziyun Zhong,<sup>a</sup> Xinyun Zhai<sup>a</sup> and Yaping Du<sup>\*,a</sup>

CoF<sub>2</sub>, with a relatively high theoretical capacity (553 mA h g<sup>-1</sup>), has been attracting increasing attention in the energy storage field. However, a facile and controllable synthesis of monodispersed CoF<sub>2</sub> and CoF<sub>2</sub>-based nano-heterostructures have been rarely reported. In this direction, an eco-friendly and precisely controlled colloidal synthesis strategy to grow uniformly sized CoF<sub>2</sub> nanorods and LiF-tipped CoF<sub>2</sub>-nanorod heterostructures based on a seeded-growth method is established. The unveiled selective growth of LiF nanoparticles onto the two end tips of the CoF<sub>2</sub> nanorods is associated with the higher energy of tips, which favors the nucleation of LiF nanocrystals. Notably, it was found that LiF could protect CoF<sub>2</sub> from corrosion even after 9 months of aging. In addition, the as-obtained heterostructures were employed in supercapacitors and lithium sulfur batteries as cathode materials. The heterostructures consistently exhibited higher specific capacities than the corresponding two single components in both types of energy storage devices, making it a potential electrode material for energy storage applications.

Received 19th July 2022

Accepted 25th September 2022

DOI: 10.1039/d2sc04008e

rsc.li/chemical-science

## Introduction

Heterostructures with two or more components, which possess synergistic chemical and physical properties than individual components, have always been attracting the interest of researchers.<sup>1–4</sup> Heterojunctions within these structures may endow better electrical conductivity<sup>5</sup> and faster ionic transport, benefiting advanced applications in catalysis<sup>6–8</sup> and electrochemistry.<sup>9,10</sup> Although many studies have been conducted on the construction of heterostructures, the controlled synthesis of high-quality heterostructures through colloidal chemistry has always been challenging.<sup>11–13</sup>

It is known that the oxides, sulfides, and phosphides of cobalt have been widely used in energy storage devices and researchers have designed some efficient strategies for boosting their performance. Therefore, the development of synthesis strategies for heterostructures was taken widely as a feasible

and efficient method, including Co<sub>3</sub>O<sub>4</sub>/CoMoO<sub>4</sub> for lithium-ion batteries,<sup>14</sup> CoO/Co–Cu–S for hybrid supercapacitors,<sup>15</sup> Co<sub>9</sub>S<sub>8</sub>/CoO for lithium sulfur batteries,<sup>16</sup> CoP/Co<sub>2</sub>P for magnesium/seawater batteries,<sup>17</sup> to name a few, and the hybrid components mentioned above exhibited better performance than their single components.

In addition to the cobalt compounds mentioned above, CoF<sub>2</sub>, which was used for Li(Na)-ion batteries as a cathode material because of its relatively high theoretical capacity (553 mA h g<sup>-1</sup>), has attracted the attention of researchers recently.<sup>18–21</sup> However, the poor conductivity and short-term cycles limit its further applications. In order to improve the performance of CoF<sub>2</sub> as an electrode material, some strategies have been reported.<sup>20,22,23</sup> Evidently, it has been challenging to obtain monodispersed CoF<sub>2</sub> nanostructures according to the previously reported synthetic methods. Additionally, in these wet chemical synthesis methods, environmentally unfriendly HF and NH<sub>4</sub>F were always used as fluoride sources or TOPO (trioctyl phosphorus oxide) solvents. On the other hand, CoF<sub>2</sub> prepared through annealing encounters the size uniformity issue and cannot provide an ideal building block for further nanoarchitecturing (Table S1†). To sum up, a precise green synthesis of CoF<sub>2</sub> with functional nano-heterostructures has been rarely reported to date, which is probably associated with its strong ionic bonds, uncondusive to the construction of heterostructures.<sup>24</sup>

In this study, monodispersed CoF<sub>2</sub> nanorods were synthesized by developing a facile colloidal synthesis strategy using

<sup>a</sup>Tianjin Key Lab for Rare Earth Materials and Applications, Center for Rare Earth and Inorganic Functional Materials, Smart Sensing Interdisciplinary Science Center, School of Materials Science and Engineering, National Institute for Advanced Materials, Nankai University, Tianjin, 300350, China. E-mail: ypdu@nankai.edu.cn

<sup>b</sup>College of Chemistry, Nankai University, Tianjin, 300071, China

<sup>c</sup>Institute of New Energy Material Chemistry, School of Materials Science and Engineering, Nankai University, Tianjin, 300350, China

<sup>d</sup>Research School of Chemistry, Australian National University, Canberra 2601, ACT, Australia. E-mail: zongyou.yin@anu.edu.au

† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d2sc04008e>



trifluoroacetates as precursors. Then, the  $\text{CoF}_2$  nanorods were used as the template for constructing the  $\text{CoF}_2$ -LiF rod-like heterostructure by injecting a lithium trifluoroacetate solution. The heterostructure was composed of  $\text{CoF}_2$  nanorods and LiF nanoparticles growing at the two end tips of the rod. Importantly, the morphology of heterostructures could be controlled *via* adjusting the amount and timing of the injected lithium trifluoroacetate solution. The formation mechanism of heterostructures was investigated in detail, and it was found that the LiF particles preferentially grew at the tips of  $\text{CoF}_2$  nanorods because of the higher energy compared with that of the body. Furthermore, the heterostructure was applied in supercapacitors and lithium sulfur batteries and consistently manifested a higher specific capacity than the two single components, making it a potential electrode material for energy storage devices.

## Results and discussion

The synthesis process of rod-like  $\text{CoF}_2$  and  $\text{CoF}_2$ -LiF heterostructures is shown in Scheme 1. 1 mmol of  $\text{Co}(\text{TFA})_2$  (cobalt trifluoroacetate) was dissolved in 3 mL of OA (oleic acid), 3 mL of OM (oleylamine) and 6 mL of ODE (1-octadecene) with vigorous stirring at room temperature to form a homogeneous solution. Then, the purple solution was heated to 120 °C and kept for 0.5 h under vacuum to remove dissolved water and oxygen from the solvents. Next, nitrogen was introduced and the solution was heated rapidly to 320 °C and maintained for 1 h to harvest  $\text{CoF}_2$  nanorods (Scheme 1a). At this moment, 2 mL (OA/OM/ODE = 0.5/0.5/1 mL) of solution B (lithium trifluoroacetate solution) was injected rapidly and reacted for 5 min to obtain  $\text{CoF}_2$ -LiF heterostructures (Scheme 1b). Scheme 1c and 1d show the TEM (transmission electron microscope) images of products obtained before and after the injection of solution B. It could be seen that particles grew at both the tips of nanorods

after injecting 2 mL of solution B with a well retained rod-like morphology for heterostructures.

The magnified TEM image of  $\text{CoF}_2$  (Fig. 1a) shows a uniform rod-like morphology with length and diameter around 180 nm and 10 nm, respectively (ESI, Fig. S1†). The self-assembly behavior of the nanorods indicates their high uniformity. Fig. 1b shows the TEM image of the products after solution B 0.05 (details shown in the Experimental section) was injected. It is evident that the products still present the rod-like morphology without other byproducts. Fig. 1c indicates that the products consist of two parts: body and tips. The dark field TEM image (Fig. 1d) further verifies the formation of heterostructures. Elemental mapping results confirm that F and Co elements co-exist at the body parts (note that Li element cannot be detected by EDS (energy-dispersive X-ray spectroscopy)). The lattice fringes were obtained from HR-TEM (high-resolution TEM) (ESI, Fig. S2†), and the interplanar distance of 0.20 nm could be ascribed to the (200) facet of LiF and 0.33 nm attributable to the (110) facet of  $\text{CoF}_2$ . Furthermore, the signal of Li could be detected through XPS (X-ray photoelectron spectroscopy), indicating the presence of Li (ESI, Fig. S3†). XRD (X-ray diffraction) patterns of samples corresponded well to the standard cards of  $\text{CoF}_2$  (JCPDS# 24-03289) and LiF (JCPDS # 04-0857), indicating that tetragonal  $\text{CoF}_2$  ( $a = b = 4.695 \text{ \AA}$ ,  $c = 3.179 \text{ \AA}$ ,  $\alpha = \beta = \gamma = 90^\circ$ ) and cubic LiF ( $a = b = c = 4.027 \text{ \AA}$ ,  $\alpha = \beta = \gamma = 90^\circ$ ) have been synthesized (Fig. 1e).

A series of condition dependent experiments were designed to uncover the growth mechanism and simultaneously optimize the synthesis process for the monodispersed  $\text{CoF}_2$  nanorods. TEM images of products obtained under different temperatures are shown in Fig. S4 (ESI†). When the reaction temperature was 260 °C, the products had no clear-defined shapes (ESI, Fig. S4a†). When the temperature increased to 300 °C, it could be observed that most of the products were nanorods but were accompanied by some small particles (ESI, Fig. S4b†). Besides, the concentration of precursors played an important role in the formation of high-quality  $\text{CoF}_2$  nanorods. When solvent



Scheme 1 Schematic of the synthesis of  $\text{CoF}_2$  nanorods and  $\text{CoF}_2$ -LiF heterostructures.

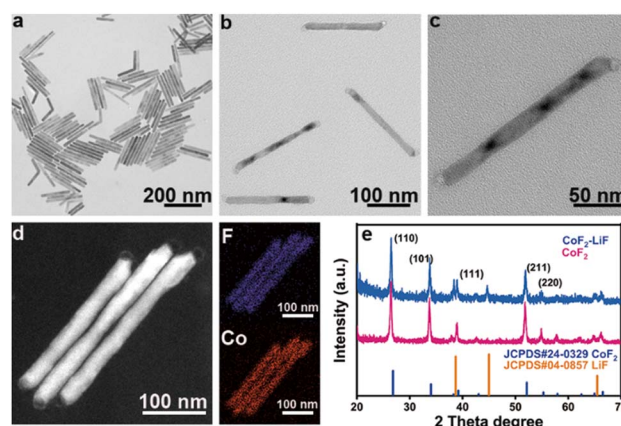


Fig. 1 TEM images of (a)  $\text{CoF}_2$  nanorods and (b and c)  $\text{CoF}_2$ -LiF heterostructures. (d) Dark field TEM images and the corresponding element mapping spectra of  $\text{CoF}_2$ -LiF heterostructures. (e) XRD patterns of  $\text{CoF}_2$  nanorods and  $\text{CoF}_2$ -LiF heterostructures.



components (OA/OM/ODE = 10/10/20 mmol) and temperature (320 °C) were fixed, using 0.5 mmol of  $\text{Co}(\text{TFA})_2$  for the reaction, irregular nanoparticles were obtained (ESI, Fig. S5a†). When 2 mmol of  $\text{Co}(\text{TFA})_2$  was added, nanorods and nanoparticles coexisted in the products (ESI, Fig. S5b†). Therefore, 1 mmol of  $\text{Co}(\text{TFA})_2$  was the optimized concentration for the preparation of uniform  $\text{CoF}_2$  nanorods in this synthesis.

Time-dependent experiments were further carried out to study the formation process of  $\text{CoF}_2$  nanorods (Fig. 2). At the beginning of the reaction (0 min), no precipitation was obtained after centrifugation, which could be due to the size of the products being too small to separate from dispersion. When the reaction time was extended to 20 min, nanorods started to appear (Fig. 2a). However, the morphology and size of products were not uniform at this stage. When solution A (details shown in the Experimental section) was kept at 320 °C for 40 min, nanorods in the products and a few irregular byproducts were obtained (Fig. 2b). Till the reaction was carried out for 60 min, the products consisted of uniform  $\text{CoF}_2$  nanorods (Fig. 2c), hinting that Ostwald ripening was well controlled.<sup>25</sup> Further, solution B 0.05 was injected into solution A rapidly and kept for 5 min, and the LiF nanoparticles grew at both the tips of  $\text{CoF}_2$  nanorods (Fig. 2d). Further, the growth time for the reaction with the injected solution B 0.05 was extended to 1 h, and there was no particle growth around the sides of  $\text{CoF}_2$  nanorods (ESI, Fig. S6†). Therefore, the amount of  $\text{Li}(\text{TFA})$  injected had a distinct effect on the morphology of products. When solution B 0.01 was injected, no LiF particles were observed on  $\text{CoF}_2$  nanorods (Fig. 2e). When solution B 0.1 (details shown in the Experimental section) was injected into solution A, LiF particles grew around  $\text{CoF}_2$  nanorods as well as at both the tips (Fig. 2f). Notably, the LiF particles were not observed growing at just one tip, indicating that the two tips possess equal opportunity to allow the nucleation-to-growth of LiF particles on the top.

What products would be obtained by injecting  $\text{Li}(\text{TFA})$  solution before the formation of  $\text{CoF}_2$  nanorods? Uniform sphere-like heterostructures were synthesized as shown in Fig. S7 (ESI†). In addition, with the increase in the concentrations of precursors, the amounts of corresponding components in heterostructures also increased (ESI, Fig. S8†). In order to

verify the good affinity between  $\text{CoF}_2$  and  $\text{LiF}$ , the  $\text{Co}(\text{TFA})_2$  solution was injected into the  $\text{Li}(\text{TFA})$  solution and kept for 30 min. In addition, the TEM image shows that the sphere-like heterostructures were still obtained (ESI, Fig. S9†). Furthermore,  $\text{CoF}_2$ - $\text{LiF}$  heterostructures were synthesized by a one-pot method instead of the hot injection method. Further, the TEM image shows that the morphology and size of products are less uniform (ESI, Fig. S10a†), which could be due to the interference of nucleation and growth between the products. Interestingly, when the given molar ratio of  $\text{Co}(\text{TFA})_2/\text{Li}(\text{TFA})$  was set at 2/1, bamboo-like heterostructures could be obtained (ESI, Fig. S10b and c†). It could be seen that the products consisted of  $\text{CoF}_2$  and  $\text{LiF}$  alternately, and the connecting part was  $\text{LiF}$ , which further indicated that the tips of  $\text{CoF}_2$  were easier to combine with  $\text{LiF}$ . Finally, the role of temperature and solvents was studied. In order to avoid the irregular nanoparticle growth at lower temperatures, the temperature was kept at 320 °C (ESI, Fig. S11†), where the heterostructure formation was dominated by thermodynamics. As observed, when the amount of either OA or OM increased, irregular heterostructures and particles were obtained. (ESI, Fig. S12a and b†). Additionally, there were nanosheets and tiny particles obtained without ODE (ESI, Fig. S12c†). To sum up, the optimized solvent components of OA/OM/ODE = 10/10/20 mmol are responsible for the mono-dispersed  $\text{CoF}_2$  nanorod growth below 320 °C.

Impressively, the morphology of heterostructures was intact even after more than 9 months of storage (Fig. 3a). In contrast, both the ends of pure  $\text{CoF}_2$  nanorods were corroded (Fig. 3b). This result strongly proved that  $\text{LiF}$  could effectively protect  $\text{CoF}_2$ . On the other hand, the two ends of pure  $\text{CoF}_2$  rods were more active than the main body because either the formation of heterostructures or the corrosion started from the two tips of nanorods. Although  $\text{LiF}$  could not be found at the tips of  $\text{CoF}_2$  nanorods from the TEM image (Fig. 3c) when solution B 0.01 was injected into solution A, the two tips of this sample were not corroded after 9 months, implying that both the tips of  $\text{CoF}_2$  nanorods were covered by  $\text{LiF}$  actually. The lattice fringes of

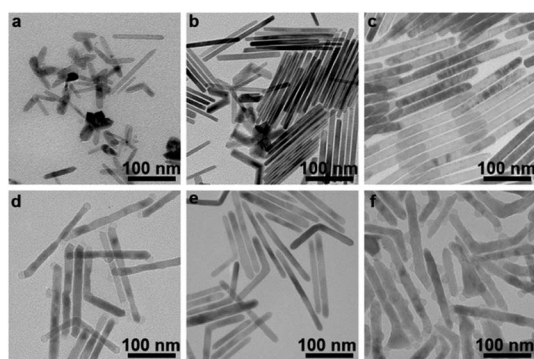


Fig. 2 TEM images of  $\text{CoF}_2$  nanorods: (a) 20 min, (b) 40 min, and (c) 60 min. TEM images of products obtained by injecting solution B (containing different amounts of  $\text{Li}(\text{TFA})$ ) into solution A: (d) 0.05 mmol, (e) 0.01 mmol, and (f) 0.1 mmol.



Fig. 3 TEM images of samples after 9 months: (a)  $\text{CoF}_2$ - $\text{LiF}$  heterostructures (0.05 mmol), (b)  $\text{CoF}_2$  nanorods, (c)  $\text{CoF}_2$ - $\text{LiF}$  heterostructures (0.01 mmol), (d) HR-TEM image of  $\text{CoF}_2$ - $\text{LiF}$  heterostructures (0.05 mmol), and (e) schematic of the formation of  $\text{CoF}_2$ - $\text{LiF}$  heterostructures.



0.33 nm correspond to that of the (110) facet of  $\text{CoF}_2$  (Fig. 3d). Based on the above mentioned results, a mechanism for the  $\text{CoF}_2$ -LiF heterostructure formation is described (Fig. 3e): the tips of  $\text{CoF}_2$  nanorods have higher energy, even when a little amount of  $\text{Li}(\text{TFA})$  solution (0.01 mmol) is injected, and LiF particles would grow at the two tips of  $\text{CoF}_2$  nanorods. This may result from the decomposition products of  $\text{Li}(\text{TFA})$  attacking the tips of  $\text{CoF}_2$  nanorods preferentially, forming nucleation sites for the LiF growth.<sup>26,27</sup> Further, the LiF nanoparticles continued to grow when there were excess amounts of  $\text{Li}(\text{TFA})$  (0.05 mmol) in the reaction system. Furthermore, the LiF particles would nucleate and grow at the body of  $\text{CoF}_2$  nanorods under the supply of more  $\text{Li}(\text{TFA})$  (increased to 0.1 mmol) after the two tip ends were covered by LiF.

The electrochemical performances of samples (LiF (details shown in the ESI, Fig. S13<sup>†</sup>),  $\text{CoF}_2$ , and  $\text{CoF}_2$ -LiF) as electrode materials were investigated using cyclic voltammetry (CV), galvanostatic charge-discharge (GCD), and electrochemical impedance spectroscopy (EIS) characterizations. The CV curves of samples measured at a scan rate of  $5 \text{ mV s}^{-1}$  are shown in Fig. 4a. Apparently, the rod-like  $\text{CoF}_2$ -LiF heterostructure electrode exhibits larger CV areas than other electrodes, indicating that the heterostructures lead to a higher capacitance.<sup>8,28</sup> The three distinct redox peaks of  $\text{CoF}_2$  nanorods and  $\text{CoF}_2$ -LiF heterostructures indicate that  $\text{CoF}_2$  undergoes two sequential electron transfer steps (from  $\text{Co}^{2+}$  to  $\text{Co}^{3+}$  to  $\text{Co}^{4+}$ ), and they possess faradaic battery-type redox capacitive

characteristics.<sup>29,30</sup> Fig. 4b shows the CV curves of the  $\text{CoF}_2$ -LiF heterostructure electrode at scan rates from 5 to  $30 \text{ mV s}^{-1}$ . All CV curves present a similar profile indicating good circulation reversibility.<sup>31</sup> Moreover, the peak currents gradually improve with the increased scan rates, which are mainly related to the polarization of electrodes.<sup>32,33</sup> The GCD curves of samples measured at  $1 \text{ A g}^{-1}$  quantify their specific capacities as shown in Fig. 4c. The nonlinear GCD curves further confirm the faradaic battery-type redox capacitive behavior, which agrees with CV curves. The  $\text{CoF}_2$ -LiF heterostructure electrode enables longer discharge time than other electrodes, corresponding to a higher capacitance. Fig. 4d demonstrates the GCD curves of  $\text{CoF}_2$ -LiF heterostructures at different current densities. The approximately symmetric charge-discharge curves verify electrochemical reversibility and good rate performance. The specific capacities calculated from the GCD curves are plotted in Fig. 4e. Notably, the  $\text{CoF}_2$ -LiF heterostructure electrode consistently exhibits higher specific capacity than the  $\text{CoF}_2$  electrode at the same current density. The specific capacity of the  $\text{CoF}_2$ -LiF heterostructures is  $32 \text{ mA h g}^{-1}$  (specific capacitance is  $200 \text{ F g}^{-1}$ ) at a current density of  $1 \text{ A g}^{-1}$ , which is comparable to that has been reported (Table S2<sup>†</sup>). The cycling stability of  $\text{CoF}_2$ -LiF and  $\text{CoF}_2$  was also evaluated as shown in Fig. S14 (ESI<sup>†</sup>), and it can be seen that at both the initial discharge time and the discharge time after 50 cycles, the heterostructure is longer than  $\text{CoF}_2$ , indicating a higher specific capacitance. By calculating the discharge time of  $\text{CoF}_2$ -LiF and  $\text{CoF}_2$ , the specific capacitance after 50 cycles was obtained as 88% and 90%, respectively. Therefore, the cycling stability of the two samples is subequal, and the heterostructures just improve the specific capacitance of  $\text{CoF}_2$ . Fig. 4f shows the EIS spectra in the Nyquist plot, which consists of a semicircle and an inclined line. The semicircle at high frequency reflects internal resistance related to the charge transfer ( $R_{ct}$ ) between the electrode surface and electrolyte.<sup>34,35</sup> The smaller semicircle indicates better electrical conductivity. According to the fitting results (Table S3<sup>†</sup>), the  $\text{CoF}_2$ -LiF heterostructure electrode has a smaller equivalent series resistance, which means that it had a higher current density at the same voltage. Combined with the reported studies, we suggest that the reason for the better performance of the heterostructures is that the unique heterogeneous architecture could enhance the charge transfer efficiency,<sup>36-38</sup> resulting in a higher specific capacity.

According to the obtained results (the  $\text{CoF}_2$ -LiF heterostructures exhibited higher specific capacity than  $\text{CoF}_2$ ), the  $\text{CoF}_2$ -LiF sample with higher electrochemical activity was further studied in the Li-S battery. To date,  $\text{CoF}_2$ , as an electrode material for fluoride-lithium and lithium-ion batteries,<sup>18-20,34,39-42</sup> has been rarely reported for Li-S batteries. In this study, the synthesized rod-like  $\text{CoF}_2$  and heterostructured  $\text{CoF}_2$ -LiF samples were used as additives in sulfur cathode materials for Li-S batteries, and the corresponding discharge-charge tests were performed at room temperature. Fig. 5 shows the galvanostatic discharge-charge curves of Li-S batteries using sulfur cathodes with  $\text{CoF}_2$ -LiF and  $\text{CoF}_2$  samples, respectively, at 0.1 C. The discharge-charge curves of the two materials are of similar trend, and the specific capacity increases gradually from the first cycle to the third cycle, which

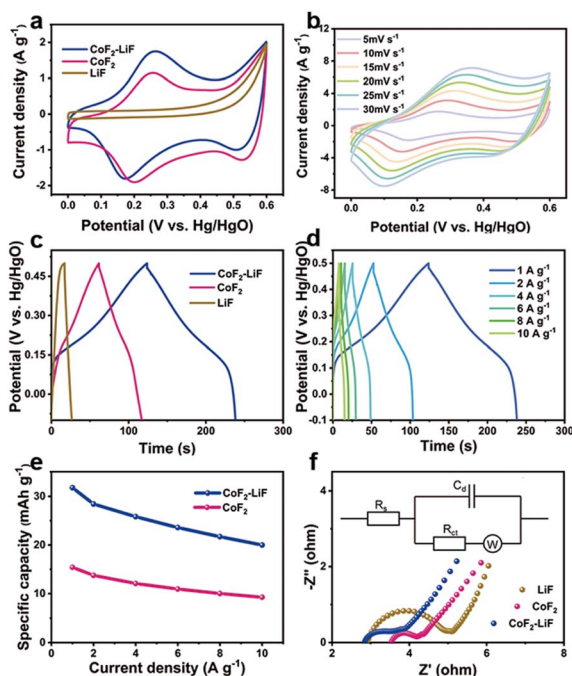


Fig. 4 (a) CV curves of LiF,  $\text{CoF}_2$ , and  $\text{CoF}_2$ -LiF at a scan rate of  $5 \text{ mV s}^{-1}$ , (b) CV curves of  $\text{CoF}_2$ -LiF heterostructures at different scan rates, (c) GCD curves of LiF,  $\text{CoF}_2$ , and  $\text{CoF}_2$ -LiF at a current density of  $1 \text{ A g}^{-1}$ , (d) GCD curves of  $\text{CoF}_2$ -LiF heterostructures at different current densities, (e) the specific capacities of  $\text{CoF}_2$  and  $\text{CoF}_2$ -LiF at different current densities, and (f) equivalent circuit diagram and Nyquist plots of LiF,  $\text{CoF}_2$ , and  $\text{CoF}_2$ -LiF.





Fig. 5 Galvanostatic charge/discharge curves at 0.1 C rate of (a)  $\text{CoF}_2\text{-LiF}$  and (b)  $\text{CoF}_2$ .

may be attributed to the activation process of the sulfur cathode. Fig. 5a shows the specific capacity of  $\text{CoF}_2\text{-LiF}$  increasing from  $585 \text{ mA h g}^{-1}$  to  $716 \text{ mA h g}^{-1}$ , which is superior to the specific capacity of  $\text{CoF}_2$  (*i.e.*, from  $445 \text{ mA h g}^{-1}$  to  $506 \text{ mA h g}^{-1}$  as shown in Fig. 5b), indicating that the  $\text{CoF}_2\text{-LiF}$  heterostructure also improves the electrochemical performance of the Li-S battery. The cycle performance and rate of Li-S battery were evaluated (ESI, Fig. S15<sup>†</sup>), and the results indicate that the specific capacitance of heterostructures is higher than that of  $\text{CoF}_2$  at 0.1 C. At different rates, the heterostructures show higher discharge capacity, which may be due to the  $\text{CoF}_2\text{-LiF}$  heterostructures accelerating the electron transport and benefiting the electrochemical reaction kinetics of Li-S cells.<sup>43–45</sup> Therefore,  $\text{CoF}_2\text{-LiF}$  heterostructures could realize better electrochemical activity. Consequently, this method is expected to expand the application of transition metal fluorides to other energy storage devices.

## Experimental

### Chemicals

Oleic acid (OA) (90%, Sigma-Aldrich), oleylamine (OM) (70%, Sigma-Aldrich), 1-octadecene (ODE) (90%, Aladdin),  $\text{Co}(\text{OH})_2$  (98%, Meryer), trifluoroacetic acid (TFA) (99%, Macklin), lithium trifluoroacetate ( $\text{Li}(\text{TFA})$ ) (97%, Macklin) *n*-hexane (AR, Aladdin), deionized water, ethanol (95%, Tianjin Concord Technology Co. Ltd), polytetrafluoroethylene (PTFE) (60%, Xiya Reagent), carbon papers (TGP-H-0600, Toray Industries, Inc.), Ketjen black (ECP 200L), and KOH (95%, Macklin) chemicals were used in for the experiments. All the reagents and solvents were used directly without further purification.

### Preparation of cobalt trifluoroacetate ( $\text{Co}(\text{TFA})_2$ )

10 mmol of  $\text{Co}(\text{OH})_2$  and 10 mL of deionized water were weighed and transferred to a 100 mL beaker. Then, the mixture was heated to  $50 \text{ }^\circ\text{C}$  with magnetic stirring. Next, 10 mL of trifluoroacetic acid was dropped and kept for 10 min to form a homogeneous dark red solution. The hot solution was filtered three times to remove impurities and dried to a pink solid at  $140 \text{ }^\circ\text{C}$ . The solid was transferred to a mortar and ground to powder. The final product was stored in a glass bottle and kept in a dryer.

### Synthesis of $\text{CoF}_2$ nanorods

1 mmol of  $\text{Co}(\text{TFA})_2$  was dissolved in OA/OM/ODE (10/10/20 mmol) in a 50 mL three-neck flask at room temperature.

Then, the mixture was heated to  $120 \text{ }^\circ\text{C}$  to remove water and oxygen under vacuum with vigorous magnetic stirring (800 rpm) for 30 min to form a dark purple homogeneous solution. The solution was heated to  $320 \text{ }^\circ\text{C}$  at a heating rate of  $20 \text{ }^\circ\text{C min}^{-1}$  and kept for 1 h under  $\text{N}_2$  atmosphere (noted as solution A). After cooling to room temperature, the products were washed with *n*-hexane/ethanol and centrifuged at 6000 rpm. Further, the precipitate was dispersed in *n*-hexane by ultrasonication. One drop of this dispersion was taken on a copper grid for morphology characterization.

### Synthesis of LiF nanoparticles

The synthetic procedure was similar to that for  $\text{CoF}_2$  nanorods except that the precursor was 1 mmol of  $\text{Li}(\text{TFA})$  and maintained at  $320 \text{ }^\circ\text{C}$  for 10 min.

### Synthesis of $\text{CoF}_2\text{-LiF}$ heterostructures

The synthetic procedure was similar to that of  $\text{CoF}_2$  nanorods except that the solution consisting of 0.01/0.05/0.1 mmol of  $\text{Li}(\text{TFA})$  and OA/OM/ODE (1.5/1.5/3 mmol) (noted as solution B 0.01/0.05/0.1, respectively) was injected to solution A and kept for 5 min.

### Electrochemical measurements

The electrochemical characterization of samples was performed in a three-electrode system with 1 M KOH as the electrolyte. A Hg/HgO electrode and platinum plate were used as the reference and counter electrodes, respectively. The working electrode was prepared by mixing samples, ketjen black, and 5% PTFE with a weight ratio of 7 : 2 : 1. The mixture was pressed onto a carbon paper ( $0.5 \text{ cm} \times 1 \text{ cm}$ ). The cyclic voltammetry (CV), galvanostatic charge–discharge (GCD), and electrochemical impedance spectroscopy (EIS) measurements were performed on an electrochemical workstation (CHI 660E) at room temperature.

### Characterization

The morphology and energy-dispersive X-ray spectroscopy (EDS) results of the samples were investigated using transmission electron microscopy (TEM) and high-resolution TEM (HR-TEM) on a HT-7800 (Hitachi, Japan) and JEM-2800 microscope (JEOL, Japan), operating at an acceleration voltage of 200 kV. The crystalline structures of the obtained samples were tested by X-ray diffraction (XRD) on a Rigaku Smart-lab X-ray diffractometer (Rigaku, Japan) with  $\text{Cu K}\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ , 20 mA, and 40 kV). X-Ray photoelectron spectroscopy (XPS) was performed using an ESCALAB 250Xi spectrometer using a monochromatic  $\text{Al K}\alpha$  X-ray source ( $h\nu = 1486.6 \text{ eV}$ ) as the excitation source (Thermal Scientific, USA).

## Conclusions

In conclusion, we developed a facile but robust method to precisely synthesize well-defined LiF-tipped  $\text{CoF}_2$ -nanorod heterostructures. The formation mechanism of heterostructures



was unveiled, showing that the selective growth of LiF nanoparticles onto the two tips of CoF<sub>2</sub> nanorods is associated with the higher energy of the tips, favoring the nucleation-to-growth of LiF nanocrystals. Unexpectedly, LiF was observed to be able to obviously protect CoF<sub>2</sub> nanorods from corrosion after long time aging. After the heterostructures were applied as supercapacitor and cathode materials for lithium sulfur batteries, they exhibited higher specific capacity than the corresponding single components, making them a potential electrode material for the next-generation energy storage. This study opens the avenue to construct advanced well-defined heterostructures and extends to various potential applications.

## Data availability

All data is available in the main text or the ESI.†

## Author contributions

The manuscript was written through the contributions of all authors. All authors have given approval for the final version of the manuscript.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

We gratefully acknowledge the support from the National Natural Science Foundation of China (21971117), the 111 Project (B18030) from China, the Beijing-Tianjin-Hebei Collaborative Innovation Project (19YFSLQY00030), the Outstanding Youth Project of Tianjin Natural Science Foundation (20JCJQC00130), the Key Project of Tianjin Natural Science Foundation (20JCZDJC00650), the Open Foundation of Guangxi Key Laboratory of Processing for Non-ferrous Metals and Featured Materials (Grant No. 2022GXYSOF07), the Functional Research Funds for the Central Universities, Nankai University (ZB19500202), Tianjin “131” Innovative Talent Team Construction Project, Tianjin Key Lab for Rare Earth Materials and Applications (ZB19500202), and the Haihe Laboratory of Sustainable Chemical Transformations for financial support.

## References

- H. Wu, O. Chen, J. Zhuang, J. Lynch, D. LaMontagne, Y. Nagaoka and Y. C. Cao, *J. Am. Chem. Soc.*, 2011, **133**, 14327–14337.
- A. M. Fagan, W. R. Jeffries, K. L. Knappenberger and R. E. Schaak, *ACS Nano*, 2021, **15**, 1378–1387.
- J. M. Hodges, J. R. Morse, J. L. Fenton, J. D. Ackerman, L. T. Alameda and R. E. Schaak, *Chem. Mater.*, 2017, **29**, 106–119.
- M. R. Buck, J. F. Bondi and R. E. Schaak, *Nat. Chem.*, 2012, **4**, 37–44.
- Z. Yu, L. Tetard, L. Zhai and J. Thomas, *Energy Environ. Sci.*, 2015, **8**, 702–730.
- W. Zhou, Z. Yin, Y. Du, X. Huang, Z. Zeng, Z. Fan, H. Liu, J. Wang and H. Zhang, *Small*, 2013, **9**, 140–147.
- H. Zhang, X. Li, A. Hähnel, V. Naumann, C. Lin, S. Azimi, S. L. Schweizer, A. W. Maijenburg and R. B. Wehrspohn, *Adv. Funct. Mater.*, 2018, **28**, 1706847.
- B. Wang, C. Tang, H. F. Wang, X. Chen, R. Cao and Q. Zhang, *Adv. Funct. Mater.*, 2019, **31**, 1805658.
- Y. Li, J. Zhang, Q. Chen, X. Xia and M. Chen, *Adv. Mater.*, 2021, **33**, 2100855.
- X. Wang, H. Li, H. Li, S. Lin, W. Ding, X. Zhu, Z. Sheng, H. Wang, X. Zhu and Y. Sun, *Adv. Funct. Mater.*, 2020, **30**, 0190302.
- S. B. Varandili, J. Huang, E. Oveisi, G. L. De Gregorio, M. Mensi, M. Strach, J. Vavra, C. Gadiyar, A. Bhowmik and R. Buonsanti, *ACS Catal.*, 2019, **9**, 5035–5046.
- J. M. Hodges, J. R. Morse, M. E. Williams and R. E. Schaak, *J. Am. Chem. Soc.*, 2015, **137**, 15493–15500.
- X. Z. Lu, C. Gu, Q. Zhang, L. Shi, S.-K. Han and G.-P. Jin, *Inorg. Chem.*, 2021, **60**, 7269–7275.
- C. He, B. Han, S. Han, Q. Xu, Z. Liang, J. Y. Xu, M. Ye, X. Liu and J. Xu, *J. Mater. Chem. A*, 2019, **7**, 26884–26892.
- W. Lu, J. Shen, P. Zhang, Y. Zhong, Y. Hu and X. W. Lou, *Angew. Chem., Int. Ed.*, 2019, **58**, 15441–15447.
- N. Wang, B. Chen, K. Qin, E. Liu, C. Shi, C. He and N. Zhao, *Nano Energy*, 2019, **60**, 332–339.
- G. Liu, M. Wang, Y. Xu, X. Wang, X. Li, J. Liu, X. Cui and L. Jiang, *J. Power Sources*, 2021, **486**, 229351.
- M. J. Armstrong, A. Panneerselvam, C. O'Regan, M. A. Morris and J. D. Holmes, *J. Mater. Chem. A*, 2013, **1**, 10667–10676.
- J. Khan, H. Ullah, M. Sajjad, A. Ali and K. H. Thebo, *Inorg. Chem. Commun.*, 2018, **98**, 132–140.
- X. Wang, W. Gu, J. T. Lee, N. Nitta, J. Benson, A. Magasinski, M. W. Schauer and G. Yushin, *Small*, 2015, **11**, 5164–5173.
- N. Zhang, X. Xiao and H. Pang, *Nanoscale Horiz.*, 2019, **4**, 99–116.
- F. Wu, V. Srot, S. Chen, M. Zhang, P. A. van Aken, Y. Wang, J. Maier and Y. Yu, *ACS Nano*, 2021, **15**, 1509–1518.
- Y. Huang, X. Li, R. Ding, D. Ying, T. Yan, Y. Huang, C. Tan, X. Sun, P. Gao and E. Liu, *Electrochim. Acta*, 2020, **329**, 135138.
- C. Coughlan, M. Ibáñez, O. Dobrozhan, A. Singh, A. Cabot and K. M. Ryan, *Chem. Rev.*, 2017, **117**, 5865–6109.
- C. B. Whitehead, S. Özkaz and R. G. Finke, *Chem. Mater.*, 2019, **31**, 7116–7132.
- Y. P. Du, Y. W. Zhang, Z. G. Yan, L. D. Sun, S. Gao and C. H. Yan, *Chem. - Asian J.*, 2007, **2**, 965–974.
- Y. P. Du, X. Sun, Y. W. Zhang, Z. G. Yan, L. D. Sun and C. H. Yan, *Cryst. Growth Des.*, 2009, **9**, 2013–2019.
- Y. Wang, Y. Liu, H. Wang, W. Liu, Y. Li, J. Zhang, H. Hou and J. Yang, *ACS Appl. Energy Mater.*, 2019, **2**, 2063–2071.
- H. Liu, J. Zhu, Z. Li, Z. Shi, J. Zhu and H. Mei, *Chem. Eng. J.*, 2021, **403**, 126325.
- C. Y. Lee, Z. Su, K. Lee, H. Tsuchiya and P. Schmuki, *Chem. Commun.*, 2014, **50**, 7067–7070.



- 31 Y. Lei, J. Li, Y. Wang, L. Gu, Y. Chang, H. Yuan and D. Xiao, *ACS Appl. Mater. Interfaces*, 2014, **6**, 1773–1780.
- 32 J. Ma, J. Xia, Z. Liang, X. Chen, Y. Du and C. H. Yan, *Small*, 2021, **17**, 2104423.
- 33 Y. Lan, H. Zhao, Y. Zong, X. Li, Y. Sun, J. Feng, Y. Wang, X. Zheng and Y. Du, *Nanoscale*, 2018, **10**, 11775–11781.
- 34 F. Wu, V. Srot, S. Chen, M. Zhang, P. A. van Aken, Y. Wang, J. Maier and Y. Yu, *ACS Nano*, 2021, **15**, 1509–1518.
- 35 C. Guo, J. Xie, J. Wang, L. Li, Z. Zhu, L. Xie, Y. Mao and W. Hu, *Adv. Energy Mater.*, 2021, **11**, 2003734.
- 36 D. Yu, B. Wu, L. Ge, L. Wu, H. Wang and T. Xu, *J. Mater. Chem. A*, 2016, **4**, 10878–10884.
- 37 P. Wang, S. Wang, X. Zhang, H. Wang, W. Duan, H. Han and X. Fan, *J. Alloys Compd.*, 2020, **819**, 153374.
- 38 P. Zhang, B. Guan, L. Yu and X. W. Lou, *Angew. Chem., Int. Ed.*, 2019, **58**, 15441–15447.
- 39 C. P. Guntlin, T. Zünd, K. V. Kravchuk, M. Wörle, M. I. Bodnarchuk and M. V. Kovalenko, *J. Mater. Chem. A*, 2017, **5**, 7383–7393.
- 40 Y. T. Teng, S. S. Pramana, J. Ding, T. Wu and R. Yazami, *Electrochim. Acta*, 2013, **107**, 301–312.
- 41 Q. Zhang, Y. T. Huang, X. Chen, A. Pan, Z. Cai, S. Liu and Y. Zhang, *J. Alloys Compd.*, 2019, **805**, 539–544.
- 42 Q. Guan, J. Cheng, X. Li, W. Ni and B. Wang, *Chin. J. Chem.*, 2017, **35**, 48–54.
- 43 J. Wang, H. Yang, Z. Chen, L. Zhang, J. Liu, P. Liang, H. Yang, X. Shen and Z. X. Shen, *Adv. Sci.*, 2018, **5**, 1800621.
- 44 Z. Li, B. Y. Guan, J. Zhang and X. W. Lou, *Joule*, 2017, **1**, 576–587.
- 45 Z. Chang, H. Dou, B. Ding, J. Wang, Y. Wang, X. Hao and D. R. MacFarlane, *J. Mater. Chem. A*, 2017, **5**, 250–257.

