

REVIEW

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Electrospun hydrogels for dynamic culture systems: advantages, progress, and opportunities

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The extracellular matrix (ECM) is a water-swollen, tissue-specific material environment in which biophysical and biochemical signals are organized and influence cell behaviors. Electrospun nanofibrous substrates have been pursued as platforms for tissue engineering and cell studies that recapitulate features of the native ECM, in particular its fibrous nature. In recent years, progress in the design of electrospun hydrogel systems has demonstrated that molecular design also enables unique studies of cellular behaviors. In comparison to the use of *hydrophobic* polymeric materials, electrospinning *hydrophilic* materials that crosslink to form hydrogels offer the potential to achieve the water-swollen, nanofibrous characteristics of endogenous ECM. Although electrospun hydrogels require an additional crosslinking step to stabilize the fibers (allowing fibers to swell with water instead of dissolving) in comparison to their hydrophobic counterparts, researchers have made significant advances in leveraging hydrogel chemistries to incorporate biochemical and dynamic functionalities within the fibers. Consequently, dynamic biophysical and biochemical properties can be engineered into hydrophilic nanofibers that would be difficult to engineer in hydrophobic systems without strategic and sometimes intensive post-processing techniques. This Review describes common methodologies to control biophysical and biochemical properties of both electrospun hydrophobic and hydrogel nanofibers, with an emphasis on highlighting recent progress using hydrogel nanofibers with engineered dynamic complexities to develop culture systems for the study of biological function, dysfunction, development, and regeneration.

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1 Introduction

The extracellular matrix (ECM) is a complex, dynamic, and tissue-specific scaffolding system that presents a myriad of biophysical and biochemical cues that influence cellular

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behaviors.^{1–4} The ECM is typically comprised of varying compositions of fibrous proteins and proteoglycans, coupled with soluble components such as growth factors;^{5–7} however, the state of this structure is constantly in flux as it is simultaneously degraded and synthesized by the resident cellular population.^{4–8} As the biophysical and biochemical attributes of the ECM at two distinct junctures are never identical, recapitulating tissue-specific milieus *in vitro* is challenging.^{5–7} To better understand cellular behaviors and processes occurring in physiologically-relevant systems, *in vitro* culture systems must continue to advance to accurately model the ECM.^{4,6,9–11}

Progress in developing more sophisticated *in vitro* culture platforms has advanced with new insights into the composition and properties of the ECM coupled with new technical capabilities to recreate its features. The heterogeneous material environment of the ECM is water-rich and nanofibrous in nature,^{1,4,12} typically comprised of single-fiber diameters on the order of tens to hundreds of nanometers (10–500 nm).^{12–16} Electrospinning is an accessible technique for depositing fibrous substrates with diameters analogous to those comprising native ECM,^{5–7} and has been established as an effective way to produce nanofibrous materials across many fields of research,^{17–21} including tissue engineering.^{22,23} Within tissue engineering and regenerative medicine, electrospun nanofibers have been applied to wound healing²⁴ and the engineering of diverse tissue types including models of cardiac,²⁵ vascular,²⁶ neural,^{27,28} and musculoskeletal²⁹ environments. In research applications addressing fundamen-

tal biological and physiological questions, electrospun substrates have also been tactically engineered to tease out cellular responses to differing environmental cues and perturbations for *in vitro* studies.^{2,3,30–32} For more information, Xue *et al.*³³ and Rahmati *et al.*³⁴ have recently published expansive reviews of the electrospinning process and extensive applications of electrospun materials.

Turning the focus from the process and applications onto the materials themselves, electrospun fibers utilized in tissue engineering applications throughout the years have been primarily comprised of hydrophobic polymers that were solubilized in organic solvents prior to electrospinning (Fig. 1). These materials were prevalent in the early waves of electrospinning due to their favorable performance in the electrospinning process and their ability to form fibrous substrates for cell culture without further stabilization steps, such as interpolymer crosslinking.^{12,35} A disadvantage of utilizing many of these hydrophobic polymers is they may lack desired cell-instructive biofunctionality in their fibrous form, and consequently require strategic chemistries to increase the bioactivity prior to seeding cells for culture.^{36,37} Furthermore, since these materials are foreign to physiological systems, it may be necessary to engineer them further to mediate biological responses *in vivo* during transplantation and degradation. There are many established methods to modify the surfaces of these hydrophobic nanofibers;^{36,37} however, a current shift towards using crosslinked polymers to develop hydrogel networks offers potential to reduce the complexity of post-proces-

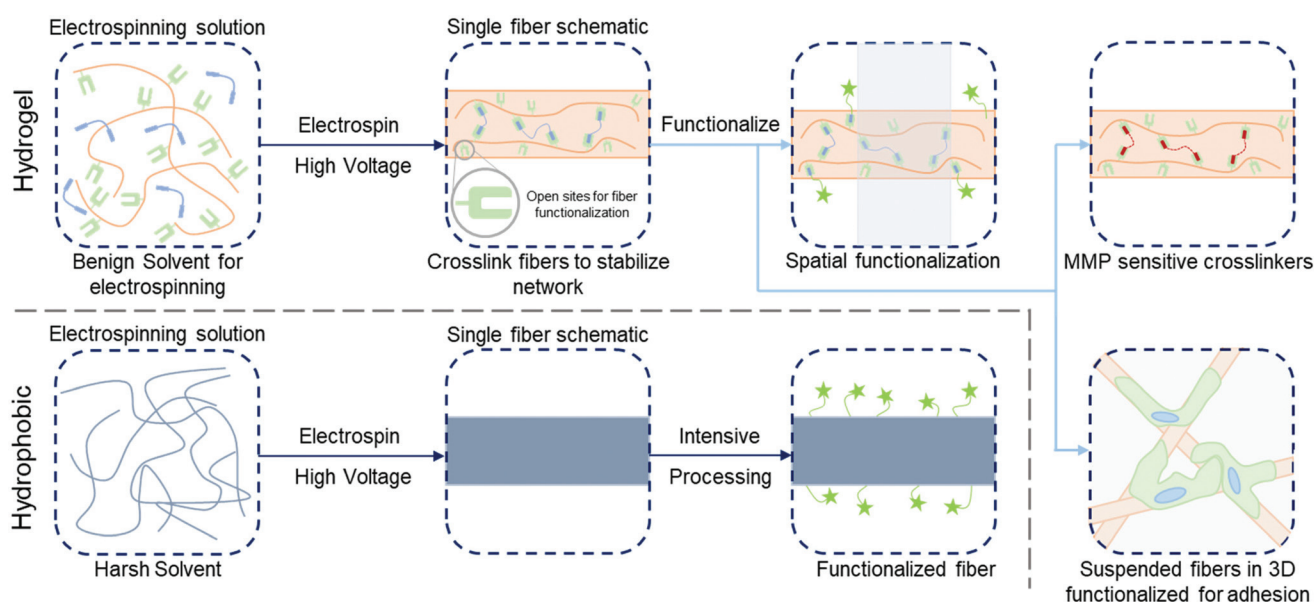


Fig. 1 Functionalization of hydrogel versus hydrophobic nanofibers. (Top, left to right): electrospinning precursor solution containing a hydrophilic polymer with a crosslinker to stabilize hydrogel nanofibers; solution is electrospun and crosslinked (e.g. with UV irradiation) with leftover sites for further functionalization; three example pathways to functionalize the fibers – spatial control over bioactivity (green stars, shaded area indicates unfunctionalized region),⁸⁶ fibers crosslinked with matrix metalloproteinase (MMP) sensitive crosslinkers for tunable degradation,⁸ suspended hydrogel fibers in a bulk gel for 3D models of the ECM.¹²³ (Bottom, left to right): electrospinning precursor solution containing hydrophobic polymer (typically in a harsh solvent); solution is electrospun and fibers are ready for processing; intensive chemical processing is typically needed for fiber functionalization.

sing (refer to Fig. 1) by drawing on the diversity of hydrogel functionalities available for modifying and controlling micro-environmental features and establishing dynamic materials.³⁸

Another advantage offered by electrospun hydrogel fibers compared to their hydrophobic material analogs is the water-swollen nature of native ECM and of natural fibers within ECM microenvironments.^{1,4,12} Furthermore, established chemistries used to modify polymeric backbones and engineer crosslinking in hydrogel fiber systems enables the facile development of functionality for controlling the biophysiochemical properties to recapitulate features of the endogenous ECM.^{1,39–41} Hydrogel systems for cell culture were originally introduced as advancements from tissue culture polystyrene,¹ and as soon as they were developed for cell culture, researchers aimed to advance the technology towards dynamic culture systems.^{4,38} Electrospun fibers are mirroring this progression first through the development of hydrogel fibers, and now in trends towards dynamic fibrous environments that allow for modeling and probing of biological processes, while also affording control over the complexity of culture systems to reconstitute natural tissue as closely as possible. Significant progress in the engineering of fibrous culture substrates has been made, with the potential for further developments in materials design to continue to advance towards recapitulating endogenous tissue.⁴²

This Review focuses on the methods developed to modify the biophysical and biochemical properties of electrospun polymers – both hydrophobic and hydrophilic – with an emphasis on the strengths provided by crosslinkable, hydrophilic polymers that form hydrogels. We further focus on the chemistries developed to modify hydrogel nanofibers to manipulate the complexity of biological systems in space and time, while additionally highlighting the advancements being made by researchers towards the development of dynamic scaffolding that effectively reconstitutes physiologically-relevant ECM. Furthermore, we also provide light commentary highlighting the advantages and associated challenges within these systems to ideally inform the next phase of advancements in nanofibrillar hydrogel design.

2 Hydrophobic polymer fibers for cell culture

The use of hydrophobic polymers has been central to the development of fibrous culture systems,⁴³ and materials commonly used include polylactic acid (PLA),^{44–47} poly(lactic-co-glycolic acid) (PLGA),⁴⁸ polycaprolactone (PCL),⁴⁹ polyethylene terephthalate (PET),⁵⁰ among many others.^{51,52} Since these materials are characteristically hydrophobic, they require non-polar organic solvents to facilitate the electrospinning process.^{25,51,53,54} Therefore – in biomedical applications – water infiltration is limited to spaces between fibers, without substantially absorbing into the polymeric matrices of the fibers themselves.⁵¹ Despite this challenge, these materials are well-suited to the electrospinning process and have seen exten-

sive use in the tissue engineering space. Part of the strength of these materials in electrospinning is that the morphological features of the resulting nanofibers can be readily tailored by simply controlling process parameters,^{12,54,55} yielding substrates with designed topographical characteristics that contribute to the biophysical properties that cells transduce. Similarly, post-electrospinning techniques have been employed to increase the bioactivity of the fibrous substrates. Since cells are heavily influenced by a combination of both biophysical and biochemical signals in their microenvironment,^{6,7} techniques have continuously progressed to introduce relevant signals to nanofibers based on these hydrophobic materials in order to influence the cells interacting with them.

2.1 Hydrophobic nanofibers enabling control over physical properties

Work aiming to engineer and alter nanofibrous topographies is driven by cellular transduction of biophysical stimuli from their microenvironments to influence signaling pathways that direct downstream phenotypic fate decisions.⁵⁶ Therefore, control over physical properties of culture systems is a critical consideration in biomedical applications including tissue engineering, regenerative medicine, and fundamental investigations into cellular processes and development. The diameters of electrospun fibers can be readily controlled through solution properties and variable parameters of the electrospinning process – in particular solution viscosity, polymer molecular weight, applied voltage, and solution flow rate.^{55,57,58} Even with this level of control, careful consideration is needed when developing fibers to match the tissue system of interest. For instance, Young's modulus of electrospun fibers exhibits an inverse relationship with fiber diameter;⁵⁹ therefore, a balance is typically needed when engineering models that replicate tissue-specific systems in the body.⁶⁰

2.1.1 Treatments for modulating fiber topography. Hydrophobic polymeric fibers are relatively robust, which allows for diverse processing techniques to further control physical and topographical properties. For example, towards engineering topography to influence cell shape and localization through contact guidance, Park and coworkers demonstrated the ability to spatially control the deposition and alignment of PLA nanofibers on polymer surfaces.⁶¹ The hydrophobicity of PLA was leveraged during the electrospinning process and an electrolyte solution of potassium chloride on the collection surface was utilized to focus the electric field during fiber collection – a process that wouldn't be possible with hydrophilic polymers.⁶¹ Moreover, from a post-processing perspective, Szczesny *et al.* heated poly(L-lactic acid) (PLLA) fibers to 85° C to induce contraction, yielding crimped fibrous substrates that recapitulated the crimped nature of tendinous tissue.⁶² Further mechanical testing showed that the crimped fibers provided a nonlinear stress-strain regime, which mirrors that seen with natural tendon tissue upon initial mechanical loading⁶² (refer to Fig. 2 Top). Towards a similar end, Chen *et al.* employed thermally-responsive materials that

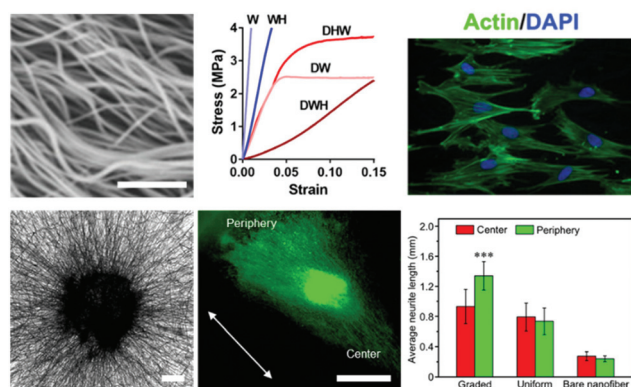


Fig. 2 Cell culture on modified hydrophobic fibrous scaffolds. (Top, left to right): crimped PLLA fibers synthesized *via* heat treatment with sacrificial fibers by Szczesny *et al.*⁶² to develop a tendinous/ligament-like tissue structure; the crimped system (DWH) exhibited a traditional non-linear stress-strain curve similar to that of native tendon/ligament tissue, whereas controls (W, WH, DHW, DW) all were unable to replicate this behavior; actin/DAPI staining of cells seeded on these crimped systems demonstrated less alignment with the fibers and reoriented significantly upon mechanical strain. Scalebar = 1 μ m. (Top) Reprinted and adapted with permission from Szczesny *et al.*, copyright 2017 American Chemical Society.⁶² (Bottom, left to right): PCL fibers aligned radially due to a novel electrospinning collection setup, scalebar = 200 μ m; Tuj-1 staining (green) of dorsal root ganglion cells shows significant neurite extension in the direction of fiber alignment (white arrow) and laminin gradient; quantification displaying average neurite length for the gradient experiments compared to controls of uniform laminin presentation and no laminin presentation. Scalebar = 1 mm, *** p < 0.001. (Bottom) Reprinted and adapted with permission from Wu *et al.*, copyright 2018 American Chemical Society.⁷⁶

shrink upon the addition of heat to crimp fibers.⁶³ The waviness in the resultant fibers improved cellular infiltration into the scaffolds, and also promoted transcriptional growth factor- β (TGF- β) expression from human mesenchymal stromal cells (hMSCs) – an important regulator in the development of connective tissue.⁶³ These examples briefly highlight the great potential hydrophobic fibers have to be tailored through modifications to the process, through post-processing, or through leveraging material properties such as thermal-responsiveness, to replicate natural tissue *in vitro*.

2.2 Hydrophobic fibers enabling modulation of biochemical properties

2.2.1 Pre-incubation (non-covalent) modifications. In addition to responding to biophysical cues in cell fate decisions, cells also integrate biochemical cues from their local microenvironment.^{7,39,64–66} Therefore, chemically modifying hydrophobic fibers that are otherwise inherently bioinert with relevant biomolecules is critical to influencing phenotypic outcomes.^{36,37} Many studies expand upon methods for introducing these biochemical cues into fibrous culture systems – often including some variation of a chemical coating as a preliminary step. For example, nonspecific adsorption of biomolecules on fibers, such as ECM-derived laminin^{27,28} and compounds contained within endothelial cell basal medium-

2,⁴⁹ supported neural and endothelial cell adhesion, respectively. Extending this pre-incubation one step further, Kador *et al.* adsorbed laminin and fibronectin onto PLA scaffolds and covalently bound Netrin-1 protein using carbodiimide (EDC/NHS) crosslinking between the carboxylic acids on laminin/fibronectin and the amines on Netrin-1.⁶⁷ Kador and coworkers also demonstrated efficacy in conjugating Netrin-1 to the laminin/fibronectin on fibers utilizing a photo-based succinimidyl-diazirine (SDA) crosslinker.⁶⁷ The immobilization of Netrin-1 on these fibrous scaffolds resulted in increased polarity of retinal ganglion cells when compared to the non-functionalized controls.⁶⁷

2.2.2 Polydopamine-based modifications. Other methods aiming to improve the biofunctionality of fibrous substrates include a preliminary step of introducing reactive chemical functionalities to fiber surfaces. Similar to the aforementioned adsorption pathways, polydopamine surface coatings, naturally inspired by the adhesiveness of mussels, allow for the presentation of catechol/quinone groups on fibers.⁶⁸ These groups can then freely react with thiols and amines of biomolecules – such as bone morphogenetic protein-2,⁶⁹ laminin,⁷⁰ or Arg-Gly-Asp (RGD) peptide motifs⁷¹ – undergoing either Schiff-base reactions or Michael additions.^{72,73}

2.2.3 High-energy surface treatments. High-energy surface treatments can also be used to introduce bioactivity. For example, Savoji and coworkers utilized plasma-polymerization to introduce a thin coating on PET nanofibers that presented reactive amine groups, which in turn supported the adhesion and subsequent proliferation of human umbilical vein endothelial cells.⁵⁰ In addition, Piai *et al.* treated PLA fibers with UV/ozone to introduce reactive oxygen groups prior to aminolysis *via* incubation in 1,6-hexamethylenediamine.⁴⁵ Chondroitin sulfate was then conjugated to the reactive amines on the PLA fibers by the aforementioned carbodiimide (EDC/NHS) crosslinking.⁴⁵ Plasma treatment has also been used in conjunction with the previously discussed polydopamine chemistry to graft another glycosaminoglycan, in this case heparin, onto polycarbonate-urethane grafts to improve bioactivity *in vivo*.⁷⁴ Moreover, Tanes *et al.*⁷⁵ and Wu *et al.*⁷⁶ both demonstrated the ability to introduce gradients of nerve growth factor (NGF)⁷⁵ and epidermal growth factor (EGF)⁷⁶ on PCL nanofibers using bovine serum albumin (BSA) as a bio-inert blocking agent. Both methods utilized oxygen plasma to functionalize the surface, prior to the sequential introduction of BSA to block open sites, then either NGF/EGF was conjugated to fibers to confer bioactivity. In the presence of both an NGF gradient and aligned fibers, dorsal root ganglion cells exhibited a preferential alignment as well as increased average length of extended neurites⁷⁵ (refer to Fig. 2 Bottom).

2.2.4 Click chemistries for biochemical modifications. Click chemistries have been explored to functionalize hydrophobic fibers with biochemical cues. Reactions that have been successfully used for controlled presentation of biomolecules include copper-catalyzed azide-alkyne cycloaddition (CuAAC) and sans metal strain-promoted azide-alkyne cycloaddition (SPAAC).³⁷ As their names reflect, CuAAC reactions require the

presentation of alkynes and azides for conjugation,⁷⁷ whereas SPAAC reactions require the presentation of strained alkynes and azides for conjugation but proceed in the absence of a copper catalyst.⁷⁸ Examples include the functionalization of PLA with an alkyne by Shi *et al.* to facilitate conjugation of an azide-presenting enzyme onto fibers through CuAAC chemistry.⁷⁹ Examples of SPAAC reactions with nanofibers include works by Smith Callahan *et al.*⁸⁰ and Zheng *et al.*⁸¹ where PLLA and PCL were functionalized with 4-dibenzo-cyclooctynol (DIBO) to provide reactive sites for conjugation of azide-containing molecules. In these works, both cell-adhesive peptides and fluorophores were conjugated to the DIBO-containing nanofibers. We refer to an excellent review by Kalaoglu-Altan *et al.* regarding 'clickable' electrospun fibers for further information on the use of bioorthogonal chemistries to modify nanofibers.³⁷

2.2.5 Summary – controlling hydrophobic nanofiber biochemical properties. Nanofibers based on hydrophobic materials have thus far been central to the development of biomedical electrospun materials and have demonstrated the progress of research in this area – becoming increasingly sophisticated, bioactive platforms with great potential in regenerative medicine. Nonetheless, these systems face certain challenges in biomedical applications that are inherent to the materials used and that can be addressed through the use of hydrogel material systems. A minor concern exists in the use of cytotoxic solvents during electrospinning to dissolve hydrophobic polymers.⁸² Although the potential to leave behind residual solvent is addressed in work with these materials, water-soluble hydrogel materials that are electrospun from aqueous solutions do not face this challenge. More significant are challenges related to advancing the biomimetic and dynamic features of electrospun fibrous systems. For example, with respect to controlling the biophysical properties of nanofibrous environments, hydrophobic systems largely afford minimal direct control over the stiffness and viscoelasticity of the resultant fibers beyond modifying solution properties prior to electrospinning. Additionally, while spatial control over the localization of biomolecules in these hydrophobic nanofibrous systems has been demonstrated through the aforementioned techniques to introduce gradients of growth factors,^{75,76} achieving complex spatiotemporal control over biochemical and biophysical features within fibrous systems remains a challenge. Progressing towards polymers used in hydrogels offers a library of existent chemistries and emerging technological advances that might be used to address many of these concerns^{38,40,41} (Fig. 1). This offers great potential to expand the possibilities within nanofibrous systems and to combine the strengths of hydrogel materials and nanofibers in engineering biomimetic environments.^{38,40,41}

3 Hydrogel nanofibers

The opportunities for increased control over the biophysical properties and spatiotemporal presentation of biochemical

functionality has been a driving factor in the progression towards electrospun *hydrogel* fibers. Hydrogel fibers build on the strengths of hydrogel materials that can be chemically modified with functional moieties – for both crosslinking and introducing biomolecules.^{1,65} These strengths allow for the precise tailoring of mechanical and chemical properties to replicate the tissue system of choice.^{1,38} Thus, hydrogel nanofibers offer the potential for greater control over fiber properties compared to their hydrophobic analogs⁸³ and the potential to provide a microenvironment that closely mirrors the water-swollen, fibrous characteristics of natural tissue.^{13–15}

3.1 Fabrication of hydrogel nanofibers

Hydrogel nanofibers are produced *via* electrospinning similarly to other variants of polymeric nanofibers. Commonly, the solution consists of the hydrophilic polymer of choice (*e.g.* hyaluronic acid (HA), poly(ethylene glycol) (PEG), or dextran), a crosslinker (for systems that require a linker molecule), a photoinitiator (for photomediated reactions), and water as a solvent.^{2,84,85} For lower molecular weight polymers, like HA and PEG, a high molecular weight polymer, typically poly(ethylene oxide), is added to increase solution viscosity and induce chain entanglements.^{32,84–86} For higher molecular weight polymers, like dextran, this is not typically needed.^{2,31,87} This solution is then typically extruded through a needle at low flow rates, where an electric field is applied to the solution. This induces a competing interaction between polymer chain entanglements within the solution and electrostatic repulsion from the voltage – which due to solution extrusion, elongates into a Taylor cone. At the point of the Taylor cone, the solution vaporizes, which causes a polymeric fiber jet to form that whips and accelerates towards the grounded collection surface.^{12,58} Following the deposition of the fibers, they must then be stabilized through some variation of crosslinking (to be described in depth below) in order to facilitate water absorption into the polymeric networks as opposed to fibers solubilizing upon hydration.^{2,84–86} Crosslinking also enables control over biophysical properties of hydrogel fibers, with degree of crosslinking directly affecting fiber parameters such as stiffness and diameter – which correlate with capacity for water swelling into the fibers.^{86,88} Once crosslinking is complete, functionalization of fibers is possible to introduce bioactivity into the fibrous hydrogel system.

3.2 Introduction to hydrogel nanofiber crosslinking and stabilization

One group of hydrogel-forming materials are natural polymers with innate biocompatibility and presentation of relevant ligands.^{89,90} For example, collagen inherently presents bioactive sites for integrin-mediated cell adhesion.¹² Other natural polymers may intrinsically interact with cells – such as hyaluronic acid (HA) (typically produced through fermentation processes¹) with CD44.^{91–93} That being said, cells tend to exhibit low adhesion to some natural polymers, like HA,

without chemical modifications to improve bioactivity.⁸⁶ Therefore, HA, as well as other polysaccharide materials such as dextran,² may need to be functionalized with bioactive molecules prior to being utilized for cell culture systems. Other hydrophilic polymers include synthetic polymers such as poly(ethylene glycol) (PEG).⁹⁴ There are a variety of established chemistries to modify the backbones of these hydrophilic polymers with pendant functional moieties that can act as sites for crosslinking and biomolecule conjugation. Therefore, modification of these polymers thereby provides significant user control over biophysical and biochemical characteristics of the nanofibers.

Unlike hydrophobic materials, as discussed previously, polymeric materials used in hydrogels are soluble in water and fibers generated by electrospinning can dissolve upon hydration without stabilization. Thus, hydrogel-based systems must generally be stabilized through some form of intermolecular crosslinking between the polymers that comprise the nanofibers. In many cases, regulation of crosslinking enables control over physical properties, as will be discussed at greater length in the next section. Naturally-derived polymers such as collagen⁹⁵ and gelatin,⁹⁶ for example, can be electrospun; however, though the native materials undergo physical crosslinking, the resultant nanofibers themselves typically are not robust enough for handling without further post-processing.^{95,96} To address this, crosslinking agents, like glutaraldehyde, have been utilized with collagen and gelatin to improve resultant mechanical properties.^{95–99} Furthermore, Kishan *et al.* developed a platform for electrospinning gelatin that crosslinks on-the-fly using a diisocyanate crosslinker to retain fiber mechanical properties.¹⁰⁰ Another effective method to stabilize collagen/gelatin-based fibers leverages carbodiimide chemistry, such as EDC/NHS crosslinking, to introduce ‘zero-length’ crosslinks.^{101–103} Chemical crosslinking has also been used to stabilize nanofibers formed from synthetic hydrophilic materials,¹⁰⁴ for example using glutaraldehyde to crosslink polyacrylamide (PA)¹⁰⁵ and poly(vinyl alcohol) (PVA).^{106–108} Glutaraldehyde as a crosslinker readily reacts with pendant groups on PA and PVA to form linkages, and offers the potential to provide user-defined control over the stiffness and swelling of resultant electrospun fibers.^{105,106}

3.3 Chemical modifications for covalent crosslinking of hydrophilic polymers

In many cases, the polymers forming the molecular backbones of these hydrogel materials are chemically modified using various strategies that enable their stabilization after electrospinning for use as fibrous hydrogel systems. Photoinitiated reactions represent a major platform for the stabilization of these hydrogel fibrous networks, and the common methodologies for photoinduced reactions utilize differing versions of the ene–ene scheme – for example through acrylate-based functional groups – and thiol–ene reactions. In the presence of light and a photoinitiator, ene–ene reactions undergo a chain-growth mechanism and form kinetic chains that crosslink the backbone polymers.¹⁰⁹ In the case of the thiol–ene reaction,

photoinitiation produces a thiyl radical, which opens and subsequently binds with an adjacent alkene enabling stoichiometric crosslinking.^{11,110–112} In addition to the crosslinking type, the degree of substitution on the polymeric backbone itself plays an important role in the regulation of downstream fiber mechanics^{113,114} – therefore, careful consideration is needed when designing the specific material system.

Many hydrophilic polymers have been modified to present pendant alkenes (using methacrylates and vinyl sulfones, for example) for crosslinking post-electrospinning. Gelatin is commonly modified with methacrylate moieties to create a material (GelMA) that can be stabilized by photoinitiated crosslinking of electrospun fibers.^{115–117} Similar chemistry has been used to modify HA,^{30,118} silk fibroin,^{119,120} and PEG.^{32,94} Dextran, another polysaccharide, can also be modified with methacrylate^{2,3,31} or vinyl sulfone⁸⁷ functional groups for crosslinking and subsequent reactions that aim at improving bioactivity. In most cases, alkene groups within nanofibers allow for anhydrous radical-induced polymerization within fibers to stabilize the polymeric networks prior to hydration.¹²¹ One of the strengths of photochemistries is the great potential for spatial control of reactions. Crosslinking, and therefore fiber stability (and ultimately mechanics), can be specified *via* selective irradiation of electrospun nanofibers through photo-masks. Sundararaghavan *et al.* used this to introduce porosity within thick fibrous substrates that would aid in cell infiltration. By masking regions of fibers during anhydrous crosslinking of methacrylated HA nanofibers, leaving them unexposed to light, regions of fibers could be selectively dissolved during hydration¹²² (see Fig. 3A).

3.4 Disadvantages and considerations when electrospinning hydrogels

Although hydrogel materials have certain advantages over their non-hydrogel counterparts, there are some associated disadvantages that need to be considered when designing these material systems for electrospinning. For example, an important consideration when using some lower molecular weight polymers, like HA and PEG, is that a carrier polymer may be required during the electrospinning process to induce chain entanglements in the solution.^{85,118} High molecular weight polymers – like poly(ethylene oxide) – may be added to the electrospinning solution to facilitate fiber formation and subsequently be washed away when the scaffolds are hydrated.¹²³ Furthermore, many biomaterials that form hydrogels are not ready for electrospinning ‘out-of-the-box’.¹ Specifically, many of the materials require chemical functionalization to introduce reactive moieties such as methacrylates,² vinyl sulfones,¹²⁴ or norbornenes⁸⁶ to the polymeric backbones. An additional consideration in using these functionalized materials is potential batch-to-batch variation in their synthesis, which may alter material properties.¹ We refer to work reviewing hydrogels for cell culture¹ for further information regarding synthesis and considerations of common hydrogel biomaterials. Finally, an inherent issue with these hydrophilic materials is the need to crosslink the fibers post-electro-

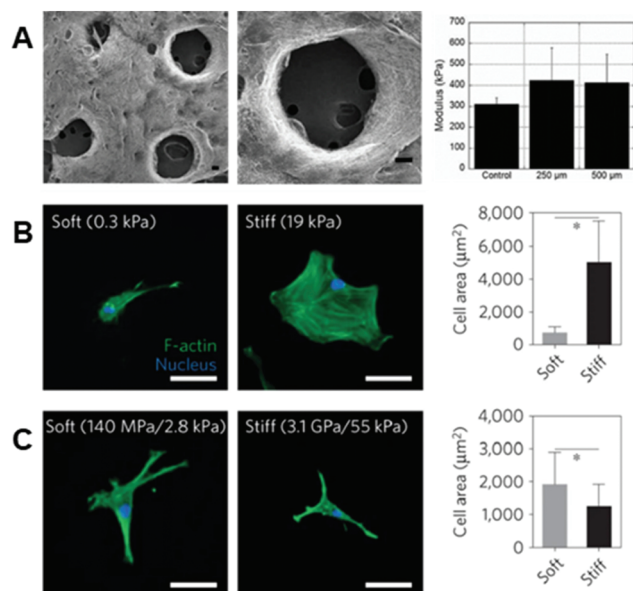


Fig. 3 Importance of fiber physical properties for cell culture. (A, left to right): SEM micrographs of MeHA fibers with user-specified photopatterned pores, zoomed in micrograph of a photopatterned pore, and a column chart displaying modulus of scaffolds – with no significant difference between scaffolds with pores and scaffolds without pores. (A) Reprinted and adapted with permission from Sundararaghavan *et al.*, copyright 2010 John Wiley and Sons;¹²² scalebars = 100 μm . (B, left to right): hMSCs show increased cell spreading on stiff hydrogels as opposed to soft hydrogels – quantified by the column chart illustrating cell area (* $p < 0.05$). (C, left to right): hMSCs demonstrate increased spreading on soft rather than stiff hydrogel fibers – quantified by the column chart showing cell area (* $p < 0.05$). These differing results emphasize the need for careful consideration when designing the biophysical properties of fibrous hydrogels for cell culture. (B) and (C) Reprinted and adapted with permission from Baker *et al.*, copyright 2015 Springer Nature;² scalebars = 50 μm .

spinning, typically prior to any further functionalization.^{2,85,86} Once the material and crosslinking strategy are chosen, however, the resultant biophysical and biochemical properties can be easily modulated – as described in the following sections. Please refer to Table 1 for a representative list of hydrogel biomaterials that have been electrospun, along with a few established methods for crosslinking and modulating the resultant biophysiochemical properties.

3.5 Hydrogel nanofibers enabling control over physical properties

As noted, the physical properties of cellular microenvironments exert strong influences over cell behaviors and phenotypes.^{125,126} In nanofibrous systems, hydrogel-based materials offer possibilities for engineering these properties, such as the mechanical and viscoelastic environments with which cells interact, within a fiber-based environment to achieve certain outcomes or interrogate biological questions.

3.5.1 Ene-ene mechanism for controlling physical properties. Within systems crosslinked *via* chain-growth polymerizations, the possibility to propagate kinetic chains after an

initial fiber-stabilizing crosslinking allows further light exposures to generate increasingly stiff fibrous networks¹¹³ as well as spatially control mechanical features. This property allows for direct-user control over resultant fiber crosslinking density, and consequently fiber stiffness, *via* irradiation duration.⁸⁷

Following the deposition and stabilization of hydrogel fibers, cell behaviors can be analyzed in *in vitro* tissue models that more closely mirror physiological features and enable experiments that assess cellular responses to perturbations of these environments. In ene-ene systems, control over mechanical properties, such as Young's modulus, has allowed cellular responses to environments of differing fiber stiffnesses to be assessed.^{2,3,30,31} For example, Baker *et al.* demonstrated, in a methacrylated-dextran system, that cell spreading behaviors on 2D stiff fibers (55 kPa, network stiffness) were inhibited in comparison to 2D soft fibers (2.8 kPa, network stiffness) – a phenomenon that is the inverse of what is seen on 2D hydrogels (Fig. 3B and C).² Baker *et al.* propose that this is due to the cells' superior ability to recruit fibers on soft substrates as opposed to stiff,² a notion that is corroborated by a computational model presented by Cao *et al.* that suggests increased focal adhesion size when matrix fibers are recruited by cells.⁴² Highlighting the complexity of mechanoresponsive cellular behaviors that can be influenced and interrogated in these systems, modulating fiber stiffness allows for design of 3D environments with high cell infiltration, combating the poor infiltration typically seen through the small pores of electrospun scaffolds.^{127–129} Interestingly, Song *et al.* demonstrated that cellular infiltration can be improved by utilizing stiffer methacrylated-hyaluronic acid (MeHA) fibers,⁸⁸ a concept that is seemingly contradictory to more cell spreading exhibited on soft fibers. This phenomenon can likely be attributed to the tendency of cells to recruit matrix fibers,^{88,130} which in turn decreases downstream pore size.⁸⁸ In fact, Song *et al.* demonstrate that on short time scales, cells invade soft fibers quickly, but then are stagnant at longer time scales – whereas cells continually invade stiff fibers across these longer time scales.⁸⁸ Furthering this, Heo *et al.* investigated the effect of nuclear stiffening as a response to matrix mechanics on cellular infiltration into these dense fibrous scaffolds.¹³¹ The result of this work demonstrated that momentary softening of the nucleus improves infiltration – suggesting that a combination of nuclear softening in conjunction with stiffer fibers can aid in cell migration into thick fibrous matrices.¹³¹

The ene-ene chain-growth polymerization is a common method for developing hydrogel fibers; however, in utilizing a chain-growth polymerization technique for crosslinking fibers and controlling mechanics, one must account for the continued growth and formation of kinetic chains in subsequent exposures to light. This additional exposure can result in increasingly stiff material environments and can cause heterogeneities leading to an inconsistent global network – an issue seen in aqueous chain-growth polymerization.^{132,133}

3.5.2 Thiol-ene mechanism for controlling physical properties. In comparison, the light-mediated thiol-ene step-

Table 1 Representative list of hydrophilic materials used to form hydrogel nanofibers with post-processing techniques

Material	Example crosslinking method(s)	Modulation of biophysiochemical properties
Fully-synthetic materials		
Polyacrylamide (PA)	Chemical: • Glutaraldehyde crosslinker ¹⁰⁵	Biochemical: • Likely adsorption-based modifications Biophysical: • Degree (extent) of crosslinking ¹⁰⁵
Poly(vinyl alcohol) (PVA)	Chemical: • Glutaraldehyde crosslinker ¹⁰⁶ • PVA composites for crosslinking ¹⁰⁷ Physical: • Controlling hydrophobicity through PVA modifications ¹⁰⁸	Biochemical: • Likely adsorption-based modifications Biophysical: • Degree (extent) of crosslinking ¹⁰⁶ or PVA modification ¹⁰⁸ • Degree of hydrolysis (<i>i.e.</i> quantity of pendant reactive groups) ¹⁰⁷
Poly(ethylene glycol) (PEG)	Chemical: • Pendant norbornenes (step-growth polymerization) ^{85,136} • Pendant methacrylates (chain-growth polymerization) ³²	Biochemical: • Adsorption-based modifications ³² • Pendant norbornenes provide sites for addition of biomolecules ◦ Light-mediated thiol-ene conjugation ⁸⁵ Biophysical: • Stiffness controlled via irradiation and crosslinker– for example: norbornenes ¹³⁶ and methacrylates ³²
Naturally-derived materials		
Collagen	Chemical: • Glutaraldehyde crosslinker ^{95,97,99} • Carbodiimide crosslinking (EDC/NHS) ¹⁰²	Biochemical: • Collagen provides natural bioactive sites for cell adhesion and interaction ⁹⁵ Biophysical: • Degree (extent) of chemical crosslinking ⁹⁷
Gelatin	Chemical: • Glutaraldehyde ⁹⁸ and diisocyanate crosslinkers ¹⁰⁰ • Carbodiimide crosslinking (EDC/NHS) ^{101,103} • Pendant methacrylates (chain-growth polymerization) ^{115–117} Physical: • Dehydrothermal crosslinking (generally weaker fibers) ⁹⁶	Biochemical: • Gelatin provides natural bioactive sites for cell adhesion and interaction ⁹⁶ Biophysical: • Degree (extent) of chemical crosslinking ⁹⁶ • Degree of chain-growth polymerization (e.g. with methacrylates) ^{115,116}
Hyaluronic acid (HA)	Chemical: • Pendant norbornenes (step-growth polymerization) ⁸⁶ • Pendant methacrylates (chain-growth polymerization) ^{30,88,118,126,144} • Pendant maleimides (chain-growth polymerization) ⁸ • Hydrazide/aldehyde proximity reactions to crosslink adjacent fibers ¹⁴⁵	Biochemical: • Pendant molecules provide sites for addition of biomolecules ◦ Michael addition: thiolated biomolecules react with pendant alkenes in basic conditions ^{8,118} ◦ Light-mediated thiol-ene conjugation ⁸⁶ Biophysical: • Stiffness also controlled via irradiation time – for example: methacrylates ^{88,122} • Stiffness within norbornene modified systems can conceivably be controlled via crosslinker added, following from Gramlich <i>et al.</i> ¹¹²
Dextran	Chemical: • Pendant methacrylates (chain-growth polymerization) ^{2,3,31} • Pendant vinyl sulfones (chain-growth polymerization) ^{87,124,146}	Biochemical: • Pendant molecules provide sites for addition of biomolecules ◦ Methacrylated heparin conjugated to free methacrylates within methacrylated-dextran fibers ⁸⁷ ◦ Michael addition: thiolated biomolecules react with pendant alkenes in basic conditions ^{2,3,31,87,124,146} Biophysical: • Stiffness also controlled via irradiation time – for example: chain-growth polymerization ²

growth polymerization offers many of the same strengths of photochemical reactions, but with increased spatiotemporal control over the formation of hydrogel networks.^{109,112} Similar to the ene–ene chemistry, hydrophilic polymers have been modified with functional groups for thiol–ene

photopolymerization. This reaction relies on a functional alkene that readily reacts with nearby thiyl radicals that are typically induced by a photoinitiator.¹¹⁰ Commonly, these polymeric backbones for electrospinning include, or are modified with, alkenes such as norbornenes^{85,86} and acry-

lates¹³⁴ – among others.¹³⁵ To crosslink the fibers, the electrospinning precursor solution must include a cross-linking molecule with multiple thiols, and after electrospinning but before hydration, fibers should be exposed to light to stabilize the fibers, similar to ene–ene chain-growth polymerization. As before, light-initiated chemistry allows spatial control over the reaction, with unexposed regions able to be dissolved away upon hydration. As mentioned, the thiol–ene reaction is advantageous because it can be designed stoichiometrically to directly control crosslinking density *via* molar ratios of reactive groups within the crosslinker relative to the polymeric backbone, with near ideal networks forming through a step-growth mechanism.¹³² The ability to control the level of crosslinking also enables residual alkenes to be preserved after crosslinking for subsequent reaction with molecules containing thiols – for example, in the addition of biomolecules,^{86,112} which will be discussed in further depth in the next section, or in introducing additional crosslinking molecules to modify mechanics with the spatiotemporal control afforded by photochemistry.

To utilize thiol–ene chemistries to engineer the mechanical environment cells interacted with, Iglesias-Echevarria *et al.* designed a coaxial electrospinning method with PCL as the core polymer for structural stability, and PEG-norbornene (PEGNB) as the sheath for tunability.¹³⁶ The PEGNB outer layer afforded control over resultant stiffness of the fibers, while also leaving behind residual norbornene groups for subsequent conjugation of thiolated RGD motifs for increased cell adhesion. The stiffness of the PEGNB sheath was modulated to investigate cellular response to differing environments. When bovine pulmonary artery endothelial cells were seeded on fibers of varying stiffnesses, higher cell infiltration and deposition of matrix materials (*e.g.* collagen, elastin) were seen on fibers with greater Young's moduli¹³⁶ – a result in line with those mentioned above by Song *et al.* utilizing a MeHA fibrous system.⁸⁸ Another interesting approach employed by Yang *et al.* involved electrospun poly((3-mercaptopropyl)methylsiloxane) (PMMS) with triallyl cyanurate (TAC) as the crosslinker.¹³⁷ PMMS has pendant thiol groups that can react with any of the alkenes on TAC to form a crosslink that stabilizes the fibers, with residual thiols available for further modification. In addition to the flexibility in the crosslinking afforded by this system, Yang *et al.* leveraged the residual thiols on TAC to conjugate a maleimide-modified poly(*N*-isopropylacrylamide) (PNIPAAm) to the fibers – exploiting the thermal-responsiveness of PNIPAAm for user-control over resultant fiber hydrophobicity.¹³⁷ In regard to physical properties, the thiol–ene reaction is a facile, powerful platform for the formation of hydrogel fibers for cell culture, providing high levels of control over the resultant fibrous scaffolds.

3.5.3 Summary – controlling hydrogel nanofiber physical properties. The physical properties of hydrogel nanofibers can be particularly well-regulated through photochemistries developed for bulk hydrogels; however, these platforms typically

yield static fibers without the inclusion of further processing for dynamic complexity. There exists potential for other chemistries, including *in situ* reactions to be expanded upon below in the section outlining dynamic fiber systems – which can perhaps be used in conjunction with the aforementioned photoinduced chemistries in dual-crosslinking systems. It is worth reiterating that while these hydrogel fiber systems allow control over physical properties that cells experience, regardless of how these fibers are crosslinked, the nanofiber diameters will increase upon fiber hydration – a phenomenon that is directly correlated with polymer hydrophilicity and crosslinking density.⁸⁸ Thus, careful balance and consideration are required when designing a hydrogel fiber system that recapitulates the physical properties of the tissue system of choice. However, designing an *in vitro* system that replicates physiological features of natural ECM requires an approach that considers both relevant biophysical and biochemical signals. Many of the crosslinking methods described above not only provide direct control over the physical properties, they can also be used to spatiotemporally incorporate desired biomolecules into the nanofibrillar environment.

3.6 Hydrogel fibers enabling modulation of biochemical properties

Within hydrogel materials, modifications such as those described above allow for spatiotemporal modulation not just of the biophysical properties, as there has been considerable progress in utilizing the same chemistries in controlling biochemical properties. Hydrogels can be designed such that the functional groups used to bind crosslinking molecules might also bind biofunctional molecules, and careful control of the crosslinking process can leave unreacted sites within the hydrogel after crosslinking to couple molecules that increase bioactivity for cellular studies.^{86,87} The ene–ene and thiol–ene reaction pathways that have been described above are also commonly utilized to introduce these biochemical signals; however, there are alternative chemistries under development that achieve similar results. We aim to provide an overview of chemistries for incorporating biomolecules into nanofibrous scaffolds based on hydrogel materials, where, in comparison to hydrophobic polymers, aqueous media might be used for all reactions.^{36,37,86}

3.6.1 Ene–ene mechanism for controlling biochemical properties. Ene–ene chain-growth, though more commonly employed in crosslinking fibers without further functionalization *via* the mechanism, can be used to introduce biochemical cues. For example, Davidson *et al.* conjugated methacrylated heparin to free vinyl sulfone groups on dextran fibers through ene–ene photopolymerization to investigate the influence of heparin presentation on resultant cell adhesion and matrix protein sequestration.⁸⁷ The addition of heparin was demonstrated to correlate with improved cell adhesion, as well as improved binding of cell-secreted fibronectin to the dextran fibers⁸⁷ (Fig. 4A). Extending the use of heparin to trap biomolecules such as the aforementioned cell-secreted

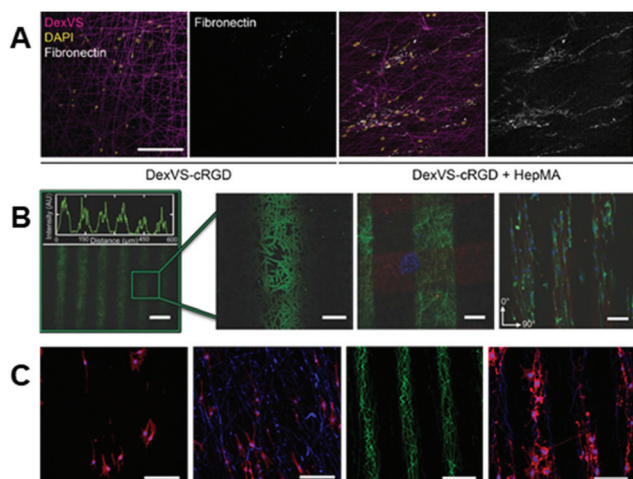


Fig. 4 Introducing biochemical cues into fibrous hydrogels. (A, left to right): Dextran-vinyl sulfone (DexVS) fibers (magenta) were seeded with human lung fibroblasts (nuclei shown in yellow) in the presence of RGD or RGD + heparin. Conjugation of RGD + heparin to DexVS fibers increased the secretion and subsequent binding of fibronectin (white) onto the fibrous matrix. (A) Reprinted and adapted with permission from Davidson *et al.*, copyright 2020 Elsevier;⁸⁷ scalebar = 200 μm . (B, left to right): spatial patterning of thiolated fluorophores onto NorHA fibers *via* thiol-ene click chemistry. Zoomed in images show high pattern fidelity, and the ability to pattern multiple biomolecules on the same scaffold – indicated by the red, green, and blue fluorophores on the fibers. The ability to pattern adhesive regions, using an RGD motif, allows for preferential cellular localization in RGD + regions that elongate in the direction of fiber alignment. (B) Reprinted and adapted with permission from Wade *et al.*, copyright 2015 John Wiley and Sons;⁸⁶ scalebars (left to right) = 100 μm , 25 μm , 100 μm , and 100 μm . (C, left to right): patterning of bioactivity on synthetic fibers using UV irradiation. Rat Schwann cells exhibited a less elongated morphology on non-bioactive substrates (far left) when compared to substrates that were activated with UV light (middle left). The use of photomasks allowed for introduction of linear bioactive regions (middle right) which promoted cell attachment over non-bioactive regions (far right). (C) Reprinted and adapted with permission from Girão *et al.* 2019;¹⁴⁷ scalebars (left to right) = 200 μm , 200 μm , 100 μm , and 100 μm .

fibronectin, Mays *et al.* conjugated methacrylated heparin to hyaluronic acid fibers to facilitate growth factor sequestration in order to promote chick dorsal root ganglia neurite length.¹³⁸

An important consideration in methods that functionalize fibers that were crosslinked *via* photoinitiated chain-growth polymerization through another photoinitiated reaction, is the effect of the subsequent reaction on kinetic chains formed during crosslinking. These kinetic chains can continue to propagate with the addition of radicals,⁸⁷ and the Young's modulus of the fibers may increase with crosslinking. To surmount this challenge, researchers may leverage the Michael-type addition reaction, where thiolated molecules bind to double bonds at slightly elevated pH, in order to incorporate functional molecules onto the pendant alkenes within these systems, avoiding further polymerization.

3.6.2 Thiol-ene (Michael Addition) for controlling biochemical properties. The Michael addition is often used to conjugate thiols to pendant alkenes in hydrogel systems.^{139–143} This chemistry allows for homogenous conjugation of thiolated biomolecules to fibrous networks containing alkenes.^{3,87,118} This conjugation can be calculated stoichiometrically, allowing for precise control over the level of functionalization. Therefore, this reaction can occur either pre-electrospinning, to modify polymeric materials that will be used in the electrospinning process,¹¹⁸ or after the cross-linking step that typically follows electrospinning.² For example, although HA is a naturally-occurring polymer that interacts with cells *via* the CD44 surface receptor, HA hydrogel substrates still require modification with ligands that can bind adhesive proteins on cell surfaces to improve cell adhesion.⁸⁶ Kim *et al.* used the base-catalyzed Michael addition to controllably introduce RGD motifs onto electrospun MeHA fibers, and demonstrated that higher presentations of RGD resulted in increased hMSC spreading, proliferation, and formation of focal adhesions.¹¹⁸ Furthermore, Sundararaghavan and Burdick were able to introduce gradients of RGD in the Z direction into dense fibrous substrates using a novel electrospinning setup that deposited unmodified MeHA and high-RGD-modified MeHA at varying flow rates.¹⁴⁴ The thiol-Michael addition is a powerful and versatile method to introduce controlled densities of biomolecules into fibrous hydrogel systems; however, due to the requirement of a basic pH for the reaction to proceed, there is minimal spatial control over the presentation of these molecules,^{2,3,30,87,118} as materials that are undergoing modification are often uniformly immersed into a basic buffer containing the thiolated molecule of interest. For spatially controlled addition of bioactivity into fibrous systems, the radical-induced thiol-ene conjugation is preferable.

3.6.3 Thiol-ene (radical induced) for controlling biochemical properties. Due to the inherent complexity of natural ECM,^{5–7} as well as the desire – in many experiments – to study cellular responses to differential signals in their microenvironments, the ability to tightly control the heterogeneity of biochemical functionalization of *in vitro* tissue culture scaffolds is desired. The radical-induced coupling of thiolated molecules onto pendant alkenes of hydrogel fibers allows for the precise localization of bioactive molecules that control cellular behaviors, such as adhesion, at high fidelity.^{86,135} As discussed previously, this photochemistry allows light exposure to control the positioning of these molecules, so photomasks or carefully focused light can be employed to control where coupling occurs in XY space. Wade and coworkers demonstrated the former using aligned electrospun nanofibers created from norbornene-functionalized hyaluronic acid (NorHA).⁸⁶ In this work, Wade *et al.* showed that through stoichiometric calculations, multiple thiolated peptides (in this case, red/green/blue fluorophores) can be conjugated to fibrous NorHA surfaces – indicating that multiple bioactive molecules can be controllably introduced.⁸⁶ Furthermore, using a thiolated RGD motif, Wade *et al.* demonstrated how 3T3 fibroblasts

responded to a combination of microenvironmental cues: a controlled spatial presentation of RGD on an aligned nanofibrous topography⁸⁶ (Fig. 4B). Moreover, Sharma and co-workers demonstrated the relative ease in employing this chemistry with PEG-norbornene fibers in a microarray system. This high-throughput platform allowed for investigation of multiple thiolated peptides with a multitude of cell types to probe cellular responses to differing microenvironments.⁸⁵ These results, taken together, clearly support the power of this chemistry scheme to control the biochemical cues that are necessary to incorporate into cell culture systems.

3.6.4 UV-irradiation for controlling biochemical properties.

In addition to radical-induced coupling, selective UV irradiation has been used to control localization of relevant biomolecules on hydrogel fibers. Similar to the UV functionalization of PLA nanofibers, Girão *et al.* used the block copolymer poly(ethylene oxide terephthalate)/poly(butylene terephthalate) (PEOT/PBT) to synthesize nanofibers.¹⁴⁷ This block copolymer provides a hydrophilic region (PEOT) and a brittle, hydrophobic region (PBT) – meaning the resultant fibers can absorb high percentages of water. The surfaces of these water-swollen fibers were then subsequently functionalized *via* selective UV irradiation to spatially control the introduction of reactive groups for biomolecule and cell adhesion. Biomolecules – such as fluorescein isothiocyanate (FITC)-tagged BSA – were conjugated vertically through the material in the XY plane and rat Schwann cells adhered selectively to functionalized regions¹⁴⁷ (Fig. 4C). The ability to tailor mechanical properties of the resultant fibers by modulating block lengths in the copolymer, in addition to spatial control over presentation of biochemical cues, makes this platform particularly attractive in the use for tissue engineering scaffolds.

3.6.5 Summary – controlling hydrogel nanofiber biochemical properties. Methods like those described above allow for easy and controllable incorporation of relevant biochemical signals into fibrous hydrogel tissue culture systems, and demonstrate strengths and potential of hydrogel-based nanofibrous platforms. It is of note that the thiol-ene reaction allows for calculated, stoichiometric crosslinking, leaving residual alkenes available for biomolecule conjugation,⁸⁶ although similar control might be exerted through careful regulation of other reactions. Light-based mechanisms offer strengths in enabling selective spatial specification of reactions. UV functionalization of fibers has demonstrated the potential to achieve the same end goal, albeit in hydrophobic materials,¹⁴⁷ whose properties such as biocompatibility, degradation, and amenability to modification must be carefully considered in material design. Other hydrophilic materials, such as hyaluronic acid and dextran, have strong track records in these areas, but the chemical structure and properties of the backbone polymer are predetermined.^{1,148} Regardless of the material selection and chemistry design, hydrogel fibers offer possibilities for high resolution spatial control over the heterogeneity of tissue culture platforms, and materials might easily be combined for next-generation fibrous systems.

4 Towards dynamic complexity and mimicking natural tissue

With technologies established to engineer nanofibrous substrates with specific biophysiochemical properties, it is possible to precisely control the spatial heterogeneity of biophysical and biochemical cues within the scaffolds. Because of this, there is exciting progress in the development of fibrous hydrogel systems that mimic natural tissue, with an emphasis on dynamic complexity – where properties of these systems might be designed to change or be controlled over time.

4.1 Engineering degradability into hydrogel nanofibers

Advances in the engineering of bulk hydrogels, both in 2D and 3D, have demonstrated unique strengths in this area – for example in material designs using enzymatically degradable crosslinkers to allow for physiologically-mediated remodeling of the scaffolds^{149–151} – and it follows that nanofibers based on hydrogel systems would have similar potential. The potential to engineer materials technologies established in bulk hydrogels into hydrogel-based nanofibers is illustrated by the development of electrospun HA fibers crosslinked with a protease-sensitive crosslinker,⁸ establishing enzymatic degradability based on materials first used as bulk hydrogels.¹⁵² Wade and coworkers electrospun a maleimide-functionalized HA with a crosslinker peptide that was degradable enzymatically by rhMMP-2 and Type II collagenase⁸ (Fig. 5A). The addition of this degradability into fibrous hydrogels allows for dynamic restructuring of the fibrous ECM by resident cells *via* the secretion of enzymes and subsequent deposition of new matrix proteins. Wade *et al.* furthered this work by demonstrating degradation *in vivo* – highlighting aspects important to translation in a subcutaneous implantation model.⁸

4.2 Dynamic fibers for selective molecule delivery

Dynamic properties in fibrous hydrogels are also embodied in applications that load the fibers with bioactive molecules to create temporal signaling. Temporal control over the release of chemokines or cytokines represent technologies with great potential for nanofibrous systems to influence cellular behavior and regeneration. Applications of controlled release from nanofibrous systems predominantly center on drug delivery applications, and there are several comprehensive reviews on this topic;^{55,82,153} we highlight systems here to illustrate technologies that might be applied in nanofibrous systems designed for tissue engineering and regenerative medicine.

Non-hydrogel fibers have demonstrated effectiveness in the delivery of molecules by both coating fibers^{154,155} and incorporating bioactive molecules in the precursor solution.¹⁵⁵ Ahire and coworkers adsorbed HA to the surface of poly(D,L, lactide) fibers and demonstrated a sustained, linear release of HA over time.¹⁵⁴ Xia *et al.* also showed efficacy in the sustained delivery of adsorbed vascular endothelial growth factor (VEGF) to the surface of poly(L-lactic acid) fibers that included nerve growth factor (NGF) in the core.¹⁵⁵ This two-step release

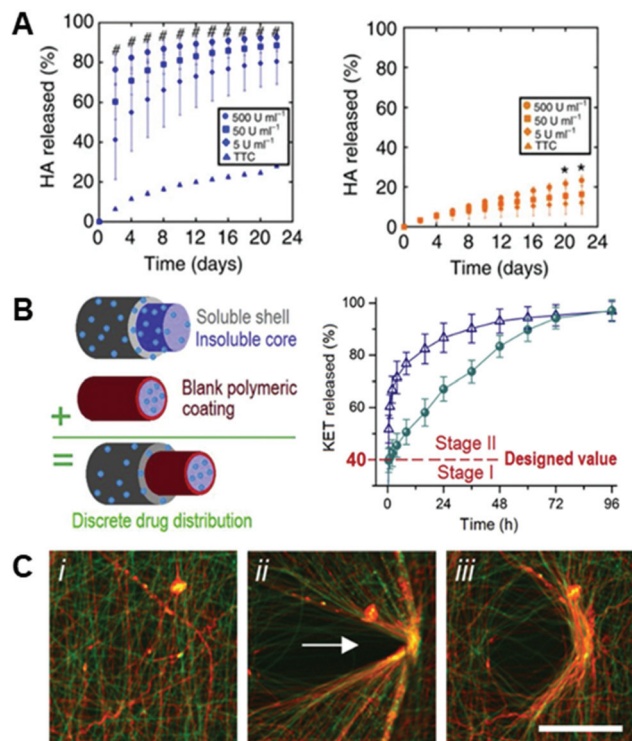


Fig. 5 Dynamic complexity in electrospun fibers. (A): HA hydrogel fibers were crosslinked with a peptide crosslinker that was susceptible to degradation via matrix metalloproteinases (MMPs). (Left): degradation of MMP-sensitive HA fibers in the presence of differing concentrations of Type II collagenase (# $p < 0.05$, for all test groups *versus* control), and (right): degradation of HA fibers crosslinked with a peptide that is *not* sensitive to Type II collagenase (* $p < 0.05$, for 500 U mL⁻¹ group *versus* control). There is a clear positive degradation effect when using an MMP-sensitive crosslinker. (A) Reprinted and adapted with permission from Wade *et al.*, copyright 2015 Springer Nature.⁸ (B): Triaxial electrospun fibers for sustained drug release. (Left): schematic of the triaxial fibers that include a polymeric coating around the innermost fiber to slow drug release. (Right): Model drug release (KET) from core-shell fibers (blue triangles) and triaxial fibers (green circles). Core-shell and tri-layered fibers both exhibited quick release past stage I (40% of release), but tri-layered fibers slowed the release throughout stage II compared to core-shell fibers – due to the polymeric coating introduced around the core. (B) Reprinted and adapted with permission from Yang *et al.*, copyright 2020 Elsevier.¹⁵⁸ (C, left to right): Hydrazide and aldehyde-functionalized NorHA fibers (i) that react to form hydrazone bonds when in contact (ii) – allowing for permanent, covalent rearrangement of fibrous scaffolds (iii). (C) Reprinted and adapted with permission from Davidson *et al.*, copyright 2019 John Wiley and Sons,¹⁴⁵ scalebars = 100 μ m.

allowed for sequential addition of biomolecules to the local environment and can, in theory, be applied to a multitude of growth/soluble factors.

Hydrogel fibers have also demonstrated promising results in the field of drug delivery. For example, Kishan and co-workers developed a platform that provides a sustained release of proteins to the local environment using different types of crosslinked gelatin fibers.¹⁵⁶ Their methacrylated gelatin system relied on traditional mass transfer for the release of a model protein incorporated within the fibers. On

the other hand, gelatin crosslinked using a diisocyanate molecule was loaded with a model protein that reacted with the gelatin backbone, and protein release in this scenario relied on gelatin degradation to free the protein from the fibers.¹⁵⁶ These two gelatin systems can be employed together to provide a tunable, sustained release of desired proteins from hydrogel fibers to support tissue growth and regeneration.

Core-shell fibers have also proven to be advantageous in the release of bioactive molecules to the adjacent environment. In the spirit of hydrogel fibers, a core-shell fibrous system was developed for the thermally-responsive release of rhodamine B.¹⁵⁷ The shell was comprised of poly-L-lactide-*co*-caprolactone (PLCL) and the core of poly(*N*-isopropylacrylamide-*co*-*N*-isopropylmethacrylamide) (P(NIPAAm-*co*-NIPMAAm)) – a thermally responsive polymer. The addition of the thermally-responsive P(NIPAAm-*co*-NIPMAAm) core allowed for a slower, more sustained release when compared to just a PLCL control.¹⁵⁷ Extending this, Yang and coworkers developed triaxial nanofibers comprised of polyvinylpyrrolidone (PVP) and cellulose acetate (CA), using ketoprofen (KET) as a model drug.¹⁵⁸ Yang *et al.* assert that the use of a tri-layered electrospun fiber yielded a more beneficial release profile initially, and the use of a CA blocking layer around the core provided a longer, more sustained release than a two layered system¹⁵⁸ (Fig. 5B). While these are select examples of the extensive work in this area,^{55,82,153} they illustrate the potential to engineer nanofibers to control release profiles and deliver important bioactive molecules relevant in cellular systems. Continuing work in designing dynamic delivery systems has direct implications for engineering temporal complexity into electrospun fibers.

4.3 Improving cell infiltration

Incorporating dynamic properties into electrospun fibers is an important consideration in developing nanofibrous scaffolds that interface with cells and natural tissue, especially in translation of regenerative materials, as touched on above with respect to controlled release. Efforts to develop dynamic fibrous structures have sought to overcome a challenge faced by electrospun fibers in implantation: small pore sizes between fibers in larger, dense mats that are of clinically relevant dimensions prevent efficient cell infiltration into the scaffolds.^{127–129} One way to surmount this challenge, in addition to the aforementioned intrafiber modifications such as enzymatically degradable crosslinks, is to spin multiple fiber types into a single substrate, where a fiber type might confer dynamic features into the substrate, such as increasing its porosity upon implantation. Specifically, water-soluble poly(ethylene oxide) (PEO) sacrificial fibers that dissolve in water, but take up space during fiber deposition and contribute to the initial structure of a larger electrospun substrate, can be co-spun with a material that is stable and persists over longer timescales.^{159–161} This method has shown to improve infiltration, without hindering cellular transduction of microenvironmental cues.¹⁶⁰ This technique has been extended to the devel-

opment of an engineered intervertebral disc, where an annulus geometry was designed with PCL fibers as the outer shell and hydrogel as the inner core.¹⁶² The addition of PEO sacrificial fibers helped increase cell infiltration into this disc model which yielded superior matrix deposition when compared to the control that did not include sacrificial fibers.¹⁶²

4.4 Molecular-level dynamic complexity

Dynamic chemistries at the molecular level also offer the potential for engineering dynamic behaviors that emerge at the scales of individual fibers and fibrous systems. Chemical crosslinking approaches that allow for fibers to rearrange in response to outside perturbations—either during assembly of structures or through interactions with cells—have been demonstrated to enable the creation of complex fibrous constructs and to allow cells to modify the physical environment they experience over time. For example, dynamic supramolecular crosslinking, where non-covalent, reversible interactions occur between complementary molecules on different polymers, can be used to assemble nanofibrous substrates and create structures with biomimetic complexity. Hyaluronic acid functionalized with methacrylates for covalent stabilization of fibers and also β -cyclodextrin (CD) (CD-MeHA) can be used to create nanofibers that form reversible bonds at interfaces with materials similarly functionalized with adamantane through supramolecular host-guest interactions.⁸⁴ CD is a cyclic host molecule with a hydrophobic core that hydrophobically interacts with guest molecules, such as adamantane (Ad) in noncovalent bonds that can be dynamically disrupted and restored.^{163–166} By designing nanofibers that present complementary functionalities on their surfaces, a nanofibrous substrate presenting CD could be adhered to another presenting Ad, offering capabilities to generate layers of aligned fibers that might be useful in cartilage or cardiac tissue engineering applications, where they might reproduce fibrous tissue structures.⁸⁴

Reversible bonds, like the Ad-CD guest-host system, have been demonstrated to introduce viscoelasticity into hydrogel tissue culture systems – allowing for cells to easily deform and remodel the local microenvironment.^{157,167,168} Nanofibrous systems with dynamic properties that enable cells to remodel their physical surroundings offer unique capabilities beyond bulk hydrogels, to observe, study, and perturb cellular behaviors through their interaction with fibrous materials. As discussed, these materials can be designed to offer ECM-like topographies as well as ECM-mimetic biophysical and biochemical features which offer cells more freedom of motion than might be achieved by encapsulating cells within a 3D hydrogel network. Towards establishing nanofibrous systems that allow dynamic, cell-responsive rearrangements of micro-environmental physical features, Davidson *et al.* used NorHA that was additionally modified with either hydrazide or aldehyde groups (NorHA-Hyd and NorHA-Ald, respectively) to dual-electrospin a fibrous blend of NorHA-Hyd and NorHA-Ald.¹⁴⁵ At the fiber surfaces, hydrazide and aldehyde functional groups reacted to form hydrazone bonds when the

two fiber types were in contact, *i.e.* an adhesive interaction^{145,152,169} (Fig. 5C). The interaction is proposed to allow cells to dynamically remodel the surrounding matrix by recruiting fibers with traction forces – with the recruited fibers subsequently reacting to preserve the structure.¹⁴⁵ Xu *et al.* also employed this chemical functionality within poly (oligoethylene glycol methacrylate) (POEGMA) fibers. POEGMA was functionalized with hydrazide/aldehyde moieties, which allowed for immediate *in situ* crosslinking following double-barrel electrospinning.¹⁶⁹ Xu *et al.* found that the hydrazide/aldehyde reaction allowed for the quick formation of crosslinks that were degradable both hydrolytically and enzymatically.¹⁶⁹

4.5 Hydrogel fibers in the third dimension

Towards increasing the dimensionality of fibrous constructs or adding fibrous features to 3D tissue models, electrospun fibers have also been employed in 3D contexts – such as dispersion into bulk hydrogels¹²⁴ and shape-shifting 3D scaffolds,¹⁷⁰ as highlighted here. The addition of fibrous networks dispersed within amorphous bulk hydrogels allows for recapitulation of the fibrillar nature of endogenous ECM, in a physiologically relevant 3D environment.⁴ For example, Matera *et al.* demonstrated increased human dermal fibroblast spreading in hydrogels with dispersed dextran fibers, as well as cellular morphological changes in a fiber density-dependent manner¹²⁴ (Fig. 6 Top). This example reinforces the influence of the biophysical signals that fibers provide within 3D cell culture systems as researchers progress towards perfecting models of ECM *in vitro*.

In an application combining electrospinning with 3D printing, Chen and coworkers demonstrated the ability to electro-

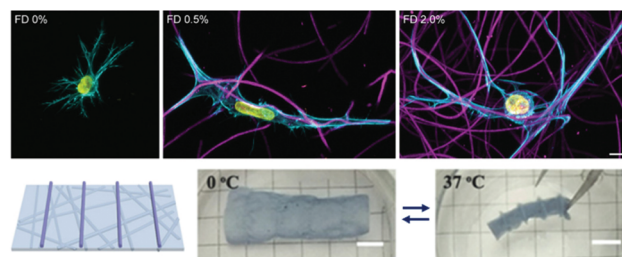


Fig. 6 Fiber suspensions in 3D hydrogels. (Top): Dispersion of DexVS fibers in 3D GelMA hydrogels. Increasing concentrations of suspended fibers (from left to right) demonstrates stark influence of fiber density on cell morphology – 0% and 2% show high levels of spread, whereas 0.5% shows a uniaxial morphology. (Top) Reprinted and adapted with permission from Matera *et al.*, copyright 2019 American Chemical Society,¹²⁴ scalebar = 10 μ m. (Bottom): P(NIPAAm-ABP) electrospun fibers with 3D printed supports. (from left to right): schematic of 3D printed supports atop of the nanofibrous P(NIPAAm-ABP) substrate; scaffold is suspended in water and adopts a relaxed conformation since the temperature is below the LCST (0° C); scaffold rolls and deforms when suspended in water with a temperature above the LCST (37° C) – thus acting as a shape-shifting hydrogel nanofiber system. (Bottom) Reprinted and adapted with permission from Chen *et al.*, copyright 2018 John Wiley and Sons,¹⁷⁰ scalebars = 5 mm.

spin poly(*N*-isopropylacrylamide) (P(NIPAAm)) hydrogel nanofibrous scaffolds that were secondarily crosslinked *via* UV light with acryloylbenzophenone (P(NIPAAm-ABP)) to form thermo-responsive mats.¹⁷⁰ Photocrosslinkable P(NIPAAm) solutions were also 3D printed onto these electrospun mats to provide rigid structure (*i.e.* trusses) to the mats. Due to P(NIPAAm)'s conformational changes above and below its lower critical solution temperature (LCST), the electrospun mats with supports exhibit shape changes upon temperature transition around the LCST due to the amount of water that is contained within the fibrous network. Below the LCST (0° C), P(NIPAAm-ABP) scaffolds demonstrated a relaxed structure; however, once the temperature was increased to above the LCST (37° C), the scaffolds rolled into shapes that were dictated by the structures 3D printed atop of the mats – hence shape-shifting nanofibrous hydrogel scaffolds (Fig. 6 Bottom).¹⁷⁰ This system demonstrates efficacy in controlling the topography of nanofibrous hydrogel culture systems and can be extended to virtually any tissue system where 3D geometric structure is of interest.

4.6 Summary – dynamic complexity and mimicking natural tissue

Work in the field continues to advance dynamic features in fibrous cell culture systems that will be central to mimicking natural tissue systems, probing fundamental biological questions, and successfully designing systems for regenerative medicine. The inclusion of protease degradable crosslinkers, dynamic remodeling, sacrificial fibers for increased cellular infiltration, and the extension towards 3D scaffolds are key progressions in the development of fiber systems. However, the field of electrospun fibers can build on progress in 2D/3D bulk hydrogel systems, and there exists clear potential for hydrogel-based nanofibers to continue to be engineered to recapitulate native physiology and control cell behaviors.

5 Next generation hydrogel fibers

As the field continues to progress towards fibrous hydrogel systems that recapture the salient features of a tissue system of interest, technology developed for engineering 2D/3D bulk hydrogels offers considerable opportunities for application in electrospun hydrogel systems. For example, expanding upon chemistries enabling dynamic degradation *via* the usage of a protease-sensitive crosslinker, chemical functionalities exist that allow directed degradation, such as photocleavable crosslinking through nitrobenzyl ether groups developed and demonstrated by the Anseth group.^{171,172} These have allowed for user-defined degradation at extremely short timescales relative to protease degradation.

Technologies that allow reversible biochemical cues to be incorporated into bulk hydrogels offer the potential for dynamic spatiotemporal control over microenvironmental features. The presentation of relevant biomolecules within the

ECM is in constant flux,^{4–7} and the ability to replicate this signaling complexity within an engineered microenvironment is critical to studying and replicating biological processes. Work that has reversibly, and repeatedly, introduced bioactive molecules into culture systems has utilized both covalent and supramolecular chemistries. Light-based approaches include nitrobenzyl ether techniques to photocleave the molecules from the scaffolds,^{10,173} and an allyl-sulfide chemistry has mediated multiple thiol-ene click reactions for incorporation and subsequent removal of desired molecules.^{11,111} These studies were conducted in PEG hydrogels, but can conceivably be applied to PEG electrospun fibers or other hydrogel fibers that are modified to support these chemistries.

Groups have also employed supramolecular chemistries to reversibly incorporate bioactive molecules in hydrogel materials. Guest-host interactions allow for self-assembly of molecules, but can be easily disrupted *via* the addition of a competing molecule.¹⁷⁴ For example, Boekhoven *et al.* utilized β -cyclodextrin as a host molecule and took advantage of differing affinities of naphthyl and adamantane to reversibly incorporate biomolecules.¹⁷⁴ To develop technology enabling greater control over these reversible interactions, oligonucleotides with toeholds have been employed for their ability to provide bioactive domains on hydrogel surfaces.¹⁷⁵ Bioactivity was removed *via* the addition of complementary oligonucleotides that took advantage of the toehold region – providing a system with defined bioactivity by cyclical addition of these oligonucleotides.¹⁷⁵ Both of these examples demonstrated the ability to control cell morphology and spreading based on the presentation of these bioactive ligands on alginate surfaces.^{174,175} Extending technologies such as these onto established hydrogel fibers would broaden opportunities to dynamically modulate complexity in water-swollen fibrous networks.

With continued progress and innovation in the materials design of fibrous hydrogel systems – and building upon exciting observations enabled by these platforms – we believe that it is inevitable that the technologies mentioned above will pave the way for platforms with increasing capabilities for recapitulating the endogenous ECM. With the growing understanding of the hydrated, fibrillar structure and function of the extracellular matrix, this progress is needed in probing fundamental physiological processes *in vitro*. As we progress, new capabilities to precisely define the biophysiochemical properties of an *in vitro* system offer opportunities for engineering biomimetic environments and controlling perturbations to homeostasis in order to understand fundamental physiological function, dysfunction, development, and regeneration. Moreover, in addition to exploring fundamental biological phenomena, technology that replicates natural tissue would enable strides towards engineered therapeutics for tissue regeneration. With applications ranging the full scale of tissue engineering – from fundamental studies to clinical translation – the development of dynamic, fibrillar hydrogels offers great potential as the field continues to develop.

Conflicts of interest

There are no conflicts to declare.

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References

- 1 S. R. Caliaari and J. A. Burdick, *Nat. Methods*, 2016, **13**, 405–414.
- 2 B. M. Baker, B. Trappmann, W. Y. Wang, M. S. Sakar, I. L. Kim, V. B. Shenoy, J. A. Burdick and C. S. Chen, *Nat. Mater.*, 2015, **14**, 1262–1268.
- 3 C. D. Davidson, W. Y. Wang, I. Zaimi, D. K. P. Jayco and B. M. Baker, *Sci. Rep.*, 2019, **9**, 12.
- 4 B. M. Baker and C. S. Chen, *J. Cell Sci.*, 2012, **125**, 3015–3024.
- 5 C. Frantz, K. M. Stewart and V. M. Weaver, *J. Cell Sci.*, 2010, **123**, 4195–4200.
- 6 R. J. Wade and J. A. Burdick, *Mater. Today*, 2012, **15**, 454–459.
- 7 M. P. Lutolf and J. A. Hubbell, *Nat. Biotechnol.*, 2005, **23**, 47–55.
- 8 R. J. Wade, E. J. Bassin, C. B. Rodell and J. A. Burdick, *Nat. Commun.*, 2015, **6**, 6639.
- 9 A. Velasco-Hogan, J. Xu and M. A. Meyers, *Adv. Mater.*, 2018, **30**(52), 1800940.
- 10 C. A. Deforest and K. S. Anseth, *Angew. Chem., Int. Ed.*, 2012, **51**, 1816–1819.
- 11 J. C. Grim, T. E. Brown, B. A. Aguado, D. A. Chapnick, A. L. Viert, X. Liu and K. S. Anseth, *ACS Cent. Sci.*, 2018, **4**, 909–916.
- 12 R. J. Wade and J. A. Burdick, *Nano Today*, 2014, **9**, 722–742.
- 13 M. D. Shoulders and R. T. Raines, *Annu. Rev. Biochem.*, 2009, **78**, 929–958.
- 14 R. O. Hynes, *Science*, 2009, **326**, 1216–1219.
- 15 L. G. Griffith and M. A. Swartz, *Mol. Cell Biol.*, 2006, **7**, 211–224.
- 16 M. M. Stevens and J. H. George, *Science*, 2005, **310**, 1135–1138.
- 17 C. Liu, P.-C. Hsu, H.-W. Lee, M. Ye, G. Zheng, N. Liu, W. Li and Y. Cui, *Nat. Commun.*, 2015, **6**, 6205.
- 18 Y. Wang, W. Li, Y. Xia, X. Jiao and D. Chen, *J. Mater. Chem. A*, 2014, **2**, 15124–15131.
- 19 T. E. Herricks, S.-H. Kim, J. Kim, D. Li, J. H. Kwak, J. W. Grate, S. H. Kim and Y. Xia, *J. Mater. Chem.*, 2005, **15**, 3241–3245.
- 20 X. Ji, P. Wang, Z. Su, G. Ma and S. Zhang, *J. Mater. Chem. B*, 2014, **2**, 181–190.
- 21 M. S. Islam, B. C. Ang, A. Andriyana and A. M. Afifi, *SN Appl. Sci.*, 2019, **1**, 1248.
- 22 H. Liu, X. Ding, G. Zhou, P. Li, X. Wei and Y. Fan, *J. Nanomater.*, 2013, 495708.
- 23 A. P. Kishan and E. M. Cosgriff-Hernandez, *J. Biomed. Mater. Res., Part A*, 2017, **105**, 2892–2905.
- 24 L. Sun, W. Gao, X. Fu, M. Shi, W. Xie, W. Zhang, F. Zhao and X. Chen, *Biomater. Sci.*, 2018, **6**, 340–349.
- 25 B. Schoen, R. Avrahami, L. Baruch, Y. Efraim, I. Goldfracht, O. Elul, T. Davidov, L. Gepstein, E. Zussman and M. Machluf, *Adv. Funct. Mater.*, 2017, **27**(34), 1700427.
- 26 S. De Valence, J. C. Tille, J. P. Giliberto, W. Mrowczynski, R. Gurny, B. H. Walpoth and M. Möller, *Acta Biomater.*, 2012, **8**, 3914–3920.
- 27 G. T. Christopherson, H. Song and H. Mao, *Biomaterials*, 2009, **30**, 556–564.
- 28 S. H. Lim, X. Y. Liu, H. Song, K. J. Yarema and H. Mao, *Biomaterials*, 2010, **31**, 9031–9039.
- 29 V. N. Chamundeswari, L. Y. Siang, Y. J. Chuah, J. S. Tan, D. A. Wang and S. C. J. Loo, *Biomed. Mater.*, 2017, **13**(1), 015019.
- 30 M. D. Davidson, K. H. Song, M. H. Lee, J. Llewellyn, Y. Du, B. M. Baker, R. G. Wells and J. A. Burdick, *ACS Biomater. Sci. Eng.*, 2019, **5**, 3899–3908.
- 31 W. Y. Wang, C. D. Davidson, D. Lin and B. M. Baker, *Nat. Commun.*, 2019, **10**, 1186.
- 32 K. Wingate, W. Bonani, Y. Tan, S. J. Bryant and W. Tan, *Acta Biomater.*, 2012, **8**, 1440–1449.
- 33 J. Xue, T. Wu, Y. Dai and Y. Xia, *Chem. Rev.*, 2019, **119**, 5298–5415.
- 34 M. Rahmati, D. K. Mills, A. M. Urbanska, M. R. Saeb, J. R. Venugopal, S. Ramakrishna and M. Mozafari, *Prog. Mater. Sci.*, DOI: 10.1016/j.pmatsci.2020.100721.
- 35 J. D. Schiffman and C. L. Schauer, *Polym. Rev.*, 2008, **48**, 317–352.
- 36 A. M. Jordan, V. Viswanath, S. Kim, J. K. Pokorski and L. T. J. Korley, *J. Mater. Chem. B*, 2016, **4**, 5958–5974.
- 37 O. I. Kalaoglu-Altan, R. Sanyal and A. Sanyal, *Polym. Chem.*, 2015, **6**, 3372–3381.
- 38 J. A. Burdick and W. L. Murphy, *Nat. Commun.*, 2012, **3**, 1269.
- 39 M. W. Tibbitt and K. S. Anseth, *Biotechnol. Bioeng.*, 2009, **103**, 655–663.
- 40 C. B. Highley, G. D. Prestwich and J. A. Burdick, *Curr. Opin. Biotechnol.*, 2016, **40**, 35–40.
- 41 J. A. Burdick and G. D. Prestwich, *Adv. Healthcare Mater.*, 2011, **23**, H41–H56.
- 42 X. Cao, E. Ban, B. M. Baker, Y. Lin, J. A. Burdick, C. S. Chen and V. B. Shenoy, *Proc. Natl. Acad. Sci. U. S. A.*, 2017, **114**, E4549–E4555.
- 43 W.-J. Li, C. T. Laurencin, E. J. Caterson, R. S. Tuan and F. K. Ko, *J. Biomed. Mater. Res.*, 2002, **60**, 613–621.
- 44 B. L.-P. Lee, H. Jeon, A. Wang, Z. Yan, J. Yu, C. Grigoropoulos and S. Li, *Acta Biomater.*, 2012, **8**, 2648–2658.

- 45 J. F. Piai, M. A. da Silva, A. Martins, A. B. Torres, S. Faria, R. L. Reis, E. C. Muniz and N. M. Neves, *Appl. Surf. Sci.*, 2017, **403**, 112–125.
- 46 I. Shabani, V. Haddadi-Asl, E. Seyedjafari and M. Soleimani, *Biochem. Biophys. Res. Commun.*, 2012, **423**, 50–54.
- 47 L. He, S. Tang, M. P. Prabhakaran, S. Liao, L. Tian, Y. Zhang, W. Xue and S. Ramakrishna, *Macromol. Biosci.*, 2013, **13**, 1601–1609.
- 48 Y.-G. Ko and O. H. Kwon, *J. Ind. Eng. Chem.*, 2020, **89**, 147–155.
- 49 S. Soliman, S. Sant, J. W. Nichol, M. Khabiry, E. Traversa and A. Khademhosseini, *J. Biomed. Mater. Res., Part A*, 2011, **96**, 566–574.
- 50 H. Savoji, A. Hadjizadeh, M. Maire, A. Ajji, M. R. Wertheimer and S. Lerouge, *Macromol. Biosci.*, 2014, **14**, 1084–1095.
- 51 X. Hu, S. Liu, G. Zhou, Y. Huang, Z. Xie and X. Jing, *J. Controlled Release*, 2014, **185**, 12–21.
- 52 C. A. Bashur, R. D. Shaffer, L. A. Dahlgren, S. A. Guelcher and A. S. Goldstein, *Tissue Eng. Part A*, 2009, **15**, 2435–2445.
- 53 A. Haider, S. Haider and I. K. Kang, *Arabian J. Chem.*, 2018, **11**, 1165–1188.
- 54 S. Agarwal, J. H. Wendorff and A. Greiner, *Polymer*, 2008, **49**, 5603–5621.
- 55 S. Chakraborty, I.-C. Liao, A. Adler and K. W. Leong, *Adv. Drug Delivery Rev.*, 2009, **61**, 1043–1054.
- 56 A. E. Miller, P. Hu and T. H. Barker, *Adv. Healthcare Mater.*, 2020, **9**, 1901445.
- 57 T. Peijs, in *Comprehensive Composite Materials II*, 2018, pp. 162–200.
- 58 Q. P. Pham, U. Sharma and A. G. Mikos, *Tissue Eng.*, 2006, **12**, 1197–1211.
- 59 S. S. Ojha, in *Electrospun Nanofibers*, 2017, pp. 239–253.
- 60 R. M. Nezarati, M. B. Eifert, D. K. Dempsey and E. Cosgriff-Hernandez, *J. Biomed. Mater. Res., Part B*, 2015, **103**, 313–323.
- 61 S. M. Park, S. Eom, D. Choi, S. J. Han, S. J. Park and D. S. Kim, *Chem. Eng. J.*, 2018, **335**, 712–719.
- 62 S. E. Szczesny, T. P. Driscoll, H.-Y. Tseng, P.-C. Liu, S.-J. Heo, R. L. Mauck and P.-H. G. Chao, *ACS Biomater. Sci. Eng.*, 2017, **3**, 2869–2876.
- 63 H. Chen, D. F. Baptista, G. Criscenti, J. Crispim, H. Fernandes, C. van Blitterswijk, R. Truckenmüller and L. Moroni, *Nanoscale*, 2019, **11**, 14312–14321.
- 64 D. Seliktar, *Science*, 2012, **336**, 1124–1129.
- 65 W. L. Murphy, T. C. McDevitt and A. J. Engler, *Nat. Mater.*, 2014, **13**, 547–557.
- 66 L. G. Griffith and G. Naughton, *Science*, 2002, **295**, 1009–1014.
- 67 K. E. Kador, H. S. Alsehli, A. N. Zindell, L. W. Lau, F. M. Andreopoulos, B. D. Watson and J. L. Goldberg, *Acta Biomater.*, 2014, **10**, 4939–4946.
- 68 H. Lee, J. Rho and P. B. Messersmith, *Adv. Mater.*, 2009, **21**, 431–434.
- 69 H. Cho, S. K. Madhurakkat Perikamana, J. Lee, J. Lee, K.-M. Lee, C. S. Shin and H. Shin, *ACS Appl. Mater. Interfaces*, 2014, **6**, 11225–11235.
- 70 N. Nazeri, R. Karimi and H. Ghanbari, *J. Biomed. Mater. Res., Part A*, 2021, **109**(2), 159–169.
- 71 Y. M. Shin, I. Jun, Y.-M. Lim, T. Rhim and H. Shin, *Macromol. Mater. Eng.*, 2013, **298**, 555–564.
- 72 M. E. Lynge, R. van der Westen, A. Postma and B. Städler, *Nanoscale*, 2011, **3**, 4916–4928.
- 73 J. H. Ryu, P. B. Messersmith and H. Lee, *ACS Appl. Mater. Interfaces*, 2018, **10**, 7523–7540.
- 74 X. Qiu, B. L.-P. Lee, X. Ning, N. Murthy, N. Dong and S. Li, *Acta Biomater.*, 2017, **51**, 138–147.
- 75 M. L. Tanes, J. Xue and Y. Xia, *J. Mater. Chem. B*, 2017, **5**, 5580–5587.
- 76 T. Wu, J. Xue, H. Li, C. Zhu, X. Mo and Y. Xia, *ACS Appl. Mater. Interfaces*, 2018, **10**, 8536–8545.
- 77 E. Haldón, M. C. Nicasio and P. J. Pérez, *Org. Biomol. Chem.*, 2015, **13**, 9528–9550.
- 78 N. E. Mbua, J. Guo, M. A. Wolfert, R. Steet and G.-J. Boons, *ChemBioChem*, 2011, **12**, 1912–1921.
- 79 Q. Shi, X. Chen, T. Lu and X. Jing, *Biomaterials*, 2008, **29**, 1118–1126.
- 80 L. A. Smith Callahan, S. Xie, I. A. Barker, J. Zheng, D. H. Reneker, A. P. Dove and M. L. Becker, *Biomaterials*, 2013, **34**, 9089–9095.
- 81 J. Zheng, S. Xie, F. Lin, G. Hua, T. Yu, D. H. Reneker and M. L. Becker, *Polym. Chem.*, 2013, **4**, 2215–2218.
- 82 V. Pillay, C. Dott, Y. E. Choonara, C. Tyagi, L. Tomar, P. Kumar, L. C. du Toit and V. M. K. Ndesendo, *J. Nanomater.*, 2013, 789289.
- 83 G. G. de Lima, S. Lyons, D. M. Devine and M. J. D. Nugent, in *Hydrogels*, Springer, Singapore, 2018, pp. 219–258.
- 84 C. B. Highley, C. B. Rodell, I. L. Kim, R. J. Wade and J. A. Burdick, *J. Mater. Chem. B*, 2015, **2**, 8110–8115.
- 85 S. Sharma, M. Floren, Y. Ding, K. R. Stenmark, W. Tan and S. J. Bryant, *Biomaterials*, 2017, **143**, 17–28.
- 86 R. J. Wade, E. J. Bassin, W. M. Gramlich and J. A. Burdick, *Adv. Mater.*, 2015, **27**, 1356–1362.
- 87 C. D. Davidson, D. K. P. Jayco, D. L. Matera, S. J. DePalma, H. L. Hiraki, W. Y. Wang and B. M. Baker, *Acta Biomater.*, 2020, **105**, 78–86.
- 88 K. H. Song, S.-J. Heo, A. P. Peredo, M. D. Davidson, R. L. Mauck and J. A. Burdick, *Adv. Healthcare Mater.*, 2019, **9**, 1901228.
- 89 L. Moroni, J. A. Burdick, C. Highley, S. J. Lee, Y. Morimoto, S. Takeuchi and J. J. Yoo, *Nat. Rev. Mater.*, 2018, **3**, 21–37.
- 90 M. Li, M. J. Mondrinos, M. R. Gandhi, F. K. Ko, A. S. Weiss and P. I. Lekes, *Biomaterials*, 2005, **26**, 5999–6008.
- 91 M. Y. Kwon, C. Wang, J. H. Galarraga, E. Puré, L. Han and J. A. Burdick, *Biomaterials*, 2019, **222**, 119451.
- 92 S. Misra, V. C. Hascall, R. R. Markwald and S. Ghatak, *Front. Immunol.*, 2015, **6**, 201.
- 93 C. Chung, M. Beecham, R. L. Mauck and J. A. Burdick, *Biomaterials*, 2009, **30**, 4287–4296.

- 94 J. J. Roberts and S. J. Bryant, *Biomaterials*, 2013, **34**, 9969–9979.
- 95 D. A. Castilla-Casadieago, H. V. Ramos-Aviles, S. Herrera-Posada, B. Calcagno, L. Loyo, J. Shipmon, A. Acevedo, A. Quintana and J. Almodovar, *Macromol. Mater. Eng.*, 2016, **301**, 1064–1075.
- 96 C. E. Campiglio, N. C. Negrini, S. Farè and L. Draghi, *Materials*, 2019, **12**, 2476.
- 97 B. S. Jha, C. E. Ayres, J. R. Bowman, T. A. Telemeco, S. A. Sell, G. L. Bowlin and D. G. Simpson, *J. Nanomater.*, 2011, 348268.
- 98 H. Aoki, H. Miyoshi and Y. Yamagata, *Polym. J.*, 2015, **47**, 267–277.
- 99 X. Zhang, K. Tang and X. Zheng, *J. Bionic Eng.*, 2016, **13**, 143–149.
- 100 A. P. Kishan, R. M. Nezarati, C. M. Radzicki, A. L. Renfro, J. L. Robinson, M. E. Whitely and E. M. Cosgriff-Hernandez, *J. Mater. Chem. B*, 2015, **3**, 7930–7938.
- 101 L. Liu, K. Kamei, M. Yoshioka, M. Nakajima, J. Li, N. Fujimoto, S. Terada, Y. Tokunaga, Y. Koyama, H. Sato, K. Hasegawa, N. Nakatsuji and Y. Chen, *Biomaterials*, 2017, **124**, 47–54.
- 102 R. L. Fischer, M. G. McCoy and S. A. Grant, *J. Mater. Sci. Mater. Med.*, 2012, **23**, 1645–1654.
- 103 Z. Ghassemi and G. Slaughter, Conf. Proc. IEEE Eng. Med. Biol. Soc., 2018, pp. 6088–6091.
- 104 R. E. Young, J. Graf, I. Miserocchi, R. M. Van Horn, M. B. Gordon, C. R. Anderson and L. S. Sefcik, *PLoS One*, 2019, **14**, 1–15.
- 105 P. Lu and Y.-L. Hsieh, *Polymer*, 2009, **50**, 3670–3679.
- 106 A. G. Destaye, C. K. Lin and C. K. Lee, *ACS Appl. Mater. Interfaces*, 2013, **5**, 4745–4752.
- 107 J. C. Park, T. Ito, K. O. Kim, K. W. Kim, B. S. Kim, M. S. Khil, H. Y. Kim and I. S. Kim, *Polym. J.*, 2010, **42**, 273–276.
- 108 C. K. Kim, B. S. Kim, F. A. Sheikh, U. S. Lee, M. S. Khil and H. Y. Kim, *Macromolecules*, 2007, **40**, 4823–4828.
- 109 E. Hui, K. I. Gimeno, G. Guan and S. R. Caliri, *Biomacromolecules*, 2019, **20**, 4126–4134.
- 110 C. E. Hoyle and C. N. Bowman, *Angew. Chem., Int. Ed.*, 2010, **49**, 1540–1573.
- 111 J. C. Grim, I. A. Marozas and K. S. Anseth, *J. Controlled Release*, 2015, **219**, 95–106.
- 112 W. M. Gramlich, I. L. Kim and J. A. Burdick, *Biomaterials*, 2013, **34**, 9803–9811.
- 113 J. L. Ifkovits and J. A. Burdick, *Tissue Eng.*, 2007, **13**, 2369–2385.
- 114 A. I. Van Den Bulcke, B. Bogdanov, N. De Rooze, E. H. Schacht, M. Cornelissen and H. Berghmans, *Biomacromolecules*, 2000, **1**, 31–38.
- 115 A. A. Aldana, L. Malatto, M. A. U. Rehman, A. R. Boccaccini and G. A. Abraham, *Nanomaterials*, 2019, **9**, 120.
- 116 X. Zhao, X. Sun, L. Yildirimer, Q. Lang, Z. Y. Lin, R. Zheng, Y. Zhang, W. Cui, N. Annabi and A. Khademhosseini, *Acta Biomater.*, 2017, **49**, 66–77.
- 117 A. O. Lobo, S. Afewerki, M. M. M. de Paula, P. Ghannadian, F. R. Marciano, Y. S. Zhang, T. J. Webster and A. Khademhosseini, *Int. J. Nanomed.*, 2018, **13**, 7891–7903.
- 118 I. L. Kim, S. Khetan, B. M. Baker, C. S. Chen and J. A. Burdick, *Biomaterials*, 2013, **34**, 5571–5580.
- 119 I. V. Bessonov, Y. A. Rochev, A. Y. Arkhipova, M. N. Kopitsyna, D. V. Bagrov, E. A. Karpushkin, T. N. Bibikova, A. M. Moysenovich, A. S. Soldatenko, I. I. Nikishin, M. S. Kotliarova, V. G. Bogush, K. V. Shaitan and M. M. Moisenovich, *Biomed. Mater.*, 2019, **14**, 034102.
- 120 S. Bin Bae, M. H. Kim and W. H. Park, *Polym. Degrad. Stab.*, 2020, **179**, 109304.
- 121 A. R. Tan, J. L. Ifkovits, B. M. Baker, D. M. Brey, R. L. Mauck and J. A. Burdick, *J. Biomed. Mater. Res., Part A*, 2008, **87**, 1034–1043.
- 122 H. G. Sundararaghavan, R. B. Metter and J. A. Burdick, *Macromol. Biosci.*, 2010, **10**, 265–270.
- 123 B. Dong, O. Arnoult, M. E. Smith and G. E. Wnek, *Macromol. Rapid Commun.*, 2009, **30**, 539–542.
- 124 D. L. Matera, W. Y. Wang, M. R. Smith, A. Shikanov and B. M. Baker, *ACS Biomater. Sci. Eng.*, 2019, **5**, 2965–2975.
- 125 M. S. Hall, F. Alisafaei, E. Ban, X. Feng, C. Y. Hui, V. B. Shenoy and M. Wu, *Proc. Natl. Acad. Sci. U. S. A.*, 2016, **113**, 14043–14048.
- 126 H. G. Sundararaghavan, R. L. Saunders, D. A. Hammer and J. A. Burdick, *Biotechnol. Bioeng.*, 2013, **110**, 1249–1254.
- 127 B. M. Baker, A. S. Nathan, G. R. Huffman and R. L. Mauck, *Osteoarthr. Cartil.*, 2009, **17**, 336–345.
- 128 F. Qu, F. Guilak and R. L. Mauck, *Nat. Rev. Rheumatol.*, 2019, **15**, 167–179.
- 129 B. M. Baker and R. L. Mauck, *Biomaterials*, 2007, **28**, 1967–1977.
- 130 A. S. Abhilash, B. M. Baker, B. Trappmann, C. S. Chen and V. B. Shenoy, *Biophys. J.*, 2014, **107**, 1829–1840.
- 131 S. J. Heo, K. H. Song, S. Thakur, L. M. Miller, X. Cao, A. P. Peredo, B. N. Seiber, F. Qu, T. P. Driscoll, V. B. Shenoy, M. Lakadamyali, J. A. Burdick and R. L. Mauck, *Sci. Adv.*, 2020, **6**, 1–13.
- 132 A. M. Kloxin, C. J. Kloxin, C. N. Bowman and K. S. Anseth, *Adv. Mater.*, 2010, **22**, 3484–3494.
- 133 S. Lin-Gibson, R. L. Jones, N. R. Washburn and F. Horkay, *Macromolecules*, 2005, **38**, 2897–2902.
- 134 K. Shanmuganathan, R. K. Sankhagowit, P. Iyer and C. J. Ellison, *Chem. Mater.*, 2011, **23**, 4726–4732.
- 135 O. I. Kalaoglu-Altan, B. Verbraeken, K. Lava, T. N. Gevrek, R. Sanyal, T. Dargaville, K. De Clerck, R. Hoogenboom and A. Sanyal, *ACS Macro Lett.*, 2016, **5**, 676–681.
- 136 M. Iglesias-Echevarria, L. Durante, R. Johnson, M. Rafuse, Y. Ding, W. Bonani, D. Maniglio and W. Tan, *Biomater. Sci.*, 2019, **7**, 3640–3651.
- 137 H. Yang, Q. Zhang, B. Lin, G. Fu, X. Zhang and L. Guo, *J. Polym. Sci., Part A: Polym. Chem.*, 2012, **50**, 4182–4190.

- 138 E. A. Mays, S. S. Kallakuri and H. G. Sundararaghavan, *J. Biomed. Mater. Res., Part A*, 2020, 1–9.
- 139 D. P. Nair, M. Podgórski, S. Chatani, T. Gong, W. Xi, C. R. Fenoli and C. N. Bowman, *Chem. Mater.*, 2014, **26**, 724–744.
- 140 C. L. Petrou, T. J. D’Ovidio, D. A. Bolukbas, S. Tas, R. D. Brown, A. Allawzi, S. Lindstedt, E. Nozik-Grayck, K. R. Stenmark, D. E. Wagner and C. M. Magin, *J. Mater. Chem. B*, 2020, **8**, 6814–6826.
- 141 B. Trappmann, B. M. Baker, W. J. Polacheck, C. K. Choi, J. A. Burdick and C. S. Chen, *Nat. Commun.*, 2017, **8**, 371.
- 142 S. P. Zustiak, H. Boukari and J. B. Leach, *Soft Matter*, 2010, **6**, 3609–3618.
- 143 K. Han, W.-N. Yin, J.-X. Fan, F.-Y. Cao and X.-Z. Zhang, *ACS Appl. Mater. Interfaces*, 2015, **7**, 23679–23684.
- 144 H. G. Sundararaghavan and J. A. Burdick, *Biomacromolecules*, 2011, **12**, 2344–2350.
- 145 M. D. Davidson, E. Ban, A. C. M. Schoonen, M.-H. Lee, M. D’Este, V. B. Shenoy and J. A. Burdick, *Adv. Mater.*, 2020, **32**(8), 1905719.
- 146 C. D. Davidson, D. K. P. Jayco, W. Y. Wang, A. Shikanov and B. M. Baker, *J. Biomech. Eng.*, 2020, **142**, 1–9.
- 147 A. F. Girão, P. Wieringa, S. C. Pinto, P. A. A. P. Marques, S. Micera, R. van Wezel, M. Ahmed, R. Truckenmueller and L. Moroni, *Front. Bioeng. Biotechnol.*, 2019, **7**, 159.
- 148 K. Saha, J. F. Pollock, D. V. Schaffer and K. E. Healy, *Curr. Opin. Chem. Biol.*, 2007, **11**, 381–387.
- 149 L. R. Nih, P. Moshayedi, I. L. Llorente, A. R. Berg, J. Cinkornpumin, W. E. Lowry, T. Segura and S. T. Carmichael, *Data in Brief*, 2017, **10**, 202–209.
- 150 L. R. Nih, E. Sideris, S. T. Carmichael and T. Segura, *Adv. Mater.*, 2017, **29**(32), 1606471.
- 151 J. L. Holloway, H. Ma, R. Rai and J. A. Burdick, *J. Controlled Release*, 2014, **191**, 63–70.
- 152 B. P. Purcell, D. Lobb, M. B. Charati, S. M. Dorsey, R. J. Wade, K. N. Zellars, H. Doviak, S. Pettaway, C. B. Logdon, J. A. Shuman, P. D. Freels, J. H. Gorman III, R. C. Gorman, F. G. Spinale and J. A. Burdick, *Nat. Mater.*, 2014, **13**, 653–661.
- 153 R. S. Bhattarai, R. D. Bachu, S. H. S. Boddu and S. Bhaduri, *Pharmaceutics*, 2019, **11**(1), DOI: 10.3390/pharmaceutics11010005.
- 154 J. J. Ahire, D. Robertson, D. P. Neveling, A. J. Van Reenen and L. M. T. Dicks, *RSC Adv.*, 2016, **6**, 34791–34796.
- 155 B. Xia and Y. Lv, *Mater. Sci. Eng. C*, 2018, **82**, 253–264.
- 156 A. Kishan, T. Walker, N. Sears, T. Wilems and E. Cosgriff-Hernandez, *J. Biomed. Mater. Res., Part A*, 2018, **106**, 1155–1164.
- 157 S. Pawłowska, C. Rinoldi, P. Nakielski, Y. Ziai, O. Urbanek, X. Li, T. A. Kowalewski, B. Ding and F. Pierini, *Adv. Mater. Interfaces*, 2020, **7**(12), 2000247.
- 158 Y. Yang, S. Chang, Y. Bai, Y. Du and D.-G. Yu, *Carbohydr. Polym.*, 2020, **243**, 116477.
- 159 M. C. Phipps, W. C. Clem, J. M. Grunda, G. A. Clines and S. L. Bellis, *Biomaterials*, 2012, **33**, 524–534.
- 160 B. M. Baker, R. P. Shah, A. M. Silverstein, J. L. Esterhai, J. A. Burdick and R. L. Mauck, *Proc. Natl. Acad. Sci. U. S. A.*, 2012, **109**, 14176–14181.
- 161 B. M. Baker, A. O. Gee, R. B. Metter, A. S. Nathan, R. A. Marklein, J. A. Burdick and R. L. Mauck, *Biomaterials*, 2008, **29**, 2348–2358.
- 162 B. G. Ashinsky, S. E. Gullbrand, E. D. Bonnevie, C. Wang, D. H. Kim, L. Han, R. L. Mauck and H. E. Smith, *Acta Biomater.*, 2020, **111**, 232–241.
- 163 M. E. Gomes, R. M. A. Domingues and R. L. Reis, *Tissue Eng., Part B*, 2017, **23**, 211–224.
- 164 C. B. Highley, C. B. Rodell and J. A. Burdick, *Adv. Mater.*, 2015, **27**, 5075–5079.
- 165 G. Sinawang, M. Osaki, Y. Takashima, H. Yamaguchi and A. Harada, *Polym. J.*, 2020, **52**, 839–859.
- 166 M. Tanaka, M. Nakahata, P. Linke and S. Kaufmann, *Polym. J.*, 2020, **52**, 861–870.
- 167 C. Loebel, R. L. Mauck and J. A. Burdick, *Nat. Mater.*, 2019, **18**, 883–891.
- 168 K. H. Vining, A. Stafford and D. J. Mooney, *Biomaterials*, 2019, **188**, 187–197.
- 169 F. Xu, H. Sheardown and T. Hoare, *Chem. Commun.*, 2016, **52**, 1451–1454.
- 170 T. Chen, H. Bakhshi, L. Liu, J. Ji and S. Agarwal, *Adv. Funct. Mater.*, 2018, **28**, 3–9.
- 171 A. M. Kloxin, A. M. Kasko, C. N. Salinas and K. S. Anseth, *Science*, 2009, **324**, 59–63.
- 172 C. A. DeForest and K. S. Anseth, *Nat. Chem.*, 2011, **3**, 925–931.
- 173 C. A. DeForest and D. A. Tirrell, *Nat. Mater.*, 2015, **14**, 523–531.
- 174 J. Boekhoven, C. M. Rubert Pérez, S. Sur, A. Worthy and S. I. Stupp, *Angew. Chem., Int. Ed.*, 2013, **52**, 12077–12080.
- 175 R. Freeman, N. Stephanopoulos, Z. Álvarez, J. A. Lewis, S. Sur, C. M. Serrano, J. Boekhoven, S. S. Lee and S. I. Stupp, *Nat. Commun.*, 2017, **8**, 15982.