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Molecular design of protein-based materials – state of the art, opportunities and challenges at the interface between materials engineering and synthetic biology

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Structural proteins like silk, squid ring teeth, elastin, collagen, or resilin, among others, are inspiring the development of new sustainable biopolymeric materials for applications including healthcare, food, soft robotics, or textiles. Furthermore, advances in the fields of soft materials and synthetic biology have a joint great potential to guide the design of novel structural proteins, despite both fields progressing mostly in a separate fashion so far. Using recombinant DNA technologies and microbial fermentations, we can design new structural proteins with monomer-level sequence control and a dispersity of ca. 1.0, based on permutations of tandem repeats derived from natural structural proteins. However, the molecular design of recombinant and repetitive structural proteins is a nontrivial task that is generally approached using low-throughput trial-and-error experimentation. Here, we review recent progress in this area, in terms of structure–function relationships and DNA synthesis technologies. We also discuss experimental and computational advances towards the establishment of rapid prototyping pipelines for this family of biopolymers. Finally, we highlight future challenges to make protein-based materials a commercially viable alternative to current fossil-based polymers.

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Design, System, Application

Structural proteins like elastin, resilin, collagen, or silk self-assemble into materials with a wide range of mechanical, structural, and dynamic properties. Inspired by this, scientists are increasingly researching structural proteins as sustainable biopolymers to manufacture materials with applications in healthcare, textiles or adhesives, to name a few. Thanks to recombinant DNA technology and synthetic biology developments, we can now generate with monomer-level precision new recombinant structural proteins that are inspired by nature, but that do not exist in nature. This allows us to create new multifunctional biopolymers that bring together dissimilar properties from natural proteins (e.g., the strength of silk, the stimuli-responsiveness of elastin, the stiffness of collagen) into a single new protein biopolymer. For these new recombinant proteins, it is hard to predict how the number and location of ordered/disordered, hydrophobic/hydrophilic, charged/uncharged, or structural/functional building blocks along their sequence affect the properties of the materials derived from them. Luckily, computational tools, such as molecular dynamics (MD) can help in unveiling sequence–property relationships, and thus accelerate the development of new recombinant proteins.

1. Introduction

Many of our everyday materials are made of polymers derived from petroleum, using toxic solvents and/or harsh

processing conditions *via* energy-intensive processes. Over 390 million metric tons of polymers are produced every year,¹ accounting for 4–6% of the annual fossil oil use.^{2,3} Fossil-based polymers also generate a vast amount of waste that is at the core of an environmental plastic pollution disaster, affecting virtually all terrestrial and aquatic ecosystems.⁴ Additionally, the growing market for polymers (with a sales turnover of over £25 billion in the UK alone in 2022 (ref. 5)) challenges our ability to meet the demands for polymeric materials while minimising waste production. This is due to the persistence of plastic pollution in nature, with degradation times exceeding centuries for some materials. Therefore, there is an urgent

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unmet need for new sustainable polymers that fulfil the current role of plastics while enabling a bio-based and circular economy.

To that end, nature is a great source of inspiration. Many natural proteins have a structural role that provides scaffolding or protection for the cells or organisms that produce those proteins. Structural proteins such as elastin, resilin, collagen, or silk can self-assemble into materials such as silkworm cocoons, the extracellular matrix, nails, squid ring teeth, or the adhesive threads of mussels.^{6–9} These materials span a wide range of mechanical and structural properties (*e.g.*, from soft to stiff, from porous to densely packed, from dynamic to static) while being based on amide bonds, much like fossil-derived polymers such as polyurethanes or nylon.¹⁰ Inspired by this, scientists are increasingly researching structural proteins as sustainable biopolymers to manufacture materials with applications in food,¹¹ medicine,¹² adhesives,¹³ energy,¹⁴ textiles¹⁵ or separation processes,¹⁶ to name a few. Additional benefits of structural proteins include sustainability, lightweight, ease of processability, degradability, stimuli responsiveness and tuneable structural/mechanical properties.¹⁷ These features are determined by nano- and mesoscale motifs (*e.g.*, β -sheets, helices, nanofibrils)¹⁸ that are encoded as short building blocks, also known as tandem repeats, in the amino acid sequence of those proteins.

Structural proteins are normally harvested from animal sources (*e.g.*, silkworm cocoons, animal tissue), but these suffer from low yields, batch-to-batch variability, presence of

contaminants, and cultural or religious concerns that limit their commercial use.¹⁸ Fortunately, advances in synthetic biology and bioprocess engineering allow us to overcome these issues and biofabricate new non-animal-derived structural proteins.¹⁹ Developments such as new DNA assembly techniques, new production hosts, or data-driven bioprocess optimisation are increasingly facilitating the biofabrication of new, high MW recombinant structural proteins with monomer-level sequence control and a dispersity of *ca.* 1.0.¹⁸ As a consequence, the yields of recombinant structural proteins (typically a few mg L⁻¹) have attained productivities above 10 g L⁻¹,²⁰ bringing industrial production of recombinant structural proteins closer to economic feasibility. Furthermore, recombinant DNA technology allows us to explore new structural proteins beyond those selected for by evolution – it is now possible to design entirely new structural proteins with properties inspired by natural ones (*e.g.*, the strength of silk, the stimuli-responsiveness of elastin, the stiffness of collagen), but that do not exist in Nature (Fig. 1). Nonetheless, this makes the molecular design space so vast as to become almost intractable *via* experimentation alone. Additionally, engineering proteins so that they self-assemble into materials with target mechanical (*e.g.*, Young's modulus, toughness) or structural (*e.g.*, mesh size, fibrillar networks) properties is a nontrivial task that is generally approached using low-throughput trial-and-error experimentation. Therefore, the design and engineering of protein-based materials would benefit from incorporating new

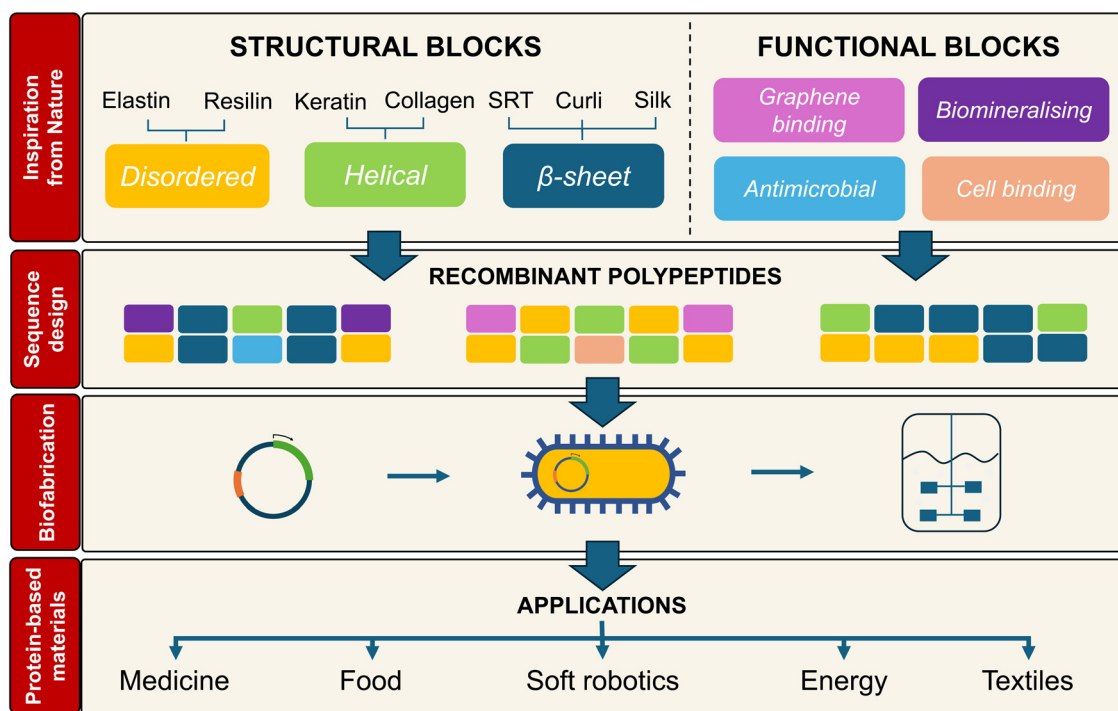


Fig. 1 Nature provides a great source of inspiration for the design of protein-based materials. Recombinant polypeptides with permutations of natural structural and functional blocks can be biofabricated to obtain new protein-based biopolymers for the manufacture of materials in multiple application areas.



computational techniques to accelerate the pace of development.

In this review, we aim to bring together two fields, synthetic biology and soft materials, that have mostly evolved independently. We first review the molecular design stage – the various protein building blocks available, their molecular-scale features, and how computational models aid in predicting *in silico* their behaviour. We then discuss synthetic biology aspects including DNA assembly methods, microbial hosts available, and the emerging field of engineered living materials (ELMs), in which the boundaries between materials and cells become blurred. Lastly, we provide an overview of the commercialisation landscape for recombinant protein biopolymers with materials applications. At the interface between biotechnology, bioprocess engineering, materials science, and computational modelling, this review aims to provide a comprehensive overview of the most recent developments in the field of microbially-produced protein-based materials.

2. Materials manufacturing from recombinant proteins

Many of the grand challenges that our society is facing, including healthcare, food and clean water supply, overuse of fossil fuels, or climate change, are intimately related to the materials that we chose to address those challenges. Fortunately, advances in engineering biology are increasingly enabling the design and engineering of new biopolymeric materials produced by microorganisms, using renewable feedstocks and sustainable biomanufacturing approaches. Recombinant synthesis of proteins is a good example of that. The genetic basis of sequence and length control in proteins, together with the manufacturing conditions applied (*e.g.*, temperature, shear, solvent), allow us to control the self-assembly of structural protein biopolymers into macroscopic materials with long-range order and bespoke mechanical, structural, or biological properties that are hard to replicate in fossil-derived materials. In the following subsections, we will review common building blocks inspired by natural structural proteins, how they influence the properties of *de novo* recombinant structural proteins, and how computational tools can accelerate the prototyping of new materials.

2.1. Nature-inspired motifs

The sequence of many natural structural proteins contains building blocks in the form of repetitive short peptides called minimal consensus repeats or tandem repeats. These repeats are usually represented by their constituent amino acids, which in this review will be noted using the single-letter code. In this section, we will review the main building blocks found in natural structural proteins, and how they determine the ability of proteins to adopt 3D conformations (*i.e.*, helices, β -sheets, β -turns, coiled-coils, or disordered regions) (Table 1) or to display a dynamic behaviour, such as stimuli responsiveness. Beyond structural roles, many proteins display other functionalities, including electrical conductivity in the pili of some bacteria or the biomineralized skeletons of diatoms,^{21,22} that have promising applications in several engineering areas. Thus, by merging structural and functional properties of natural proteins into *de novo* recombinant proteins, we can engineer new protein-based materials with properties guided from the molecular scale up. However, we are still far from fully understanding how the macroscale properties of protein-based materials are linked to design features at the molecular scale (*e.g.*, content in ordered/disordered, hydrophobic/hydrophilic, charged/uncharged, or structural/functional building blocks in the protein sequence) or the manufacturing routes used (*i.e.*, solvent-based approaches, 3D printing, thermal moulding) – the so-called sequence-processing-property relationships. This explains why, to date, new structural recombinant proteins are mainly designed through low-throughput experimental trial-and-error permutations of a few building blocks.

2.1.1. Helical and coiled coil proteins. Helices are ubiquitous structures in Nature in which a single polypeptide chain twists around itself to form a cylindrical shape. Helices have noteworthy mechanical properties, as they combine elasticity in the axial direction and bending stiffness in the radial direction.^{23,24} Furthermore, under excessive load, they can undergo a helix-to- β -sheet transition that leads to a strain-stiffening behaviour^{25,26} (Fig. 2a). Tandem repeats with a propensity towards forming helical 3D structures have been incorporated in recombinant structural proteins with the aim of enhancing the self-assembly and mechanical stability of various material formats,^{27,28} such as fibres,^{29,30} nanotubes³⁰ or cages.^{31,32}

Table 1 Summary of protein types and their corresponding amino acid sequences

Protein type	Structural block	Tandem repeat
Helical coiled coils	Collagen-like	GXY
Intrinsically disordered proteins	Elastin-like	VPGXG
	Resilin-like	GGRPSDSYGAPGGGN or GGRPSSSYGAPGQGN
β -Sheet-rich proteins	Curli	S(X) ₅ QXGXGNXA(X) ₃ Q
	Silk-like (silkworm)	GAGAGS
	Silk-like (spider)	GAGAAAAAGGAGTS
	Squid-ring-teeth-like	polyA PAAASVSTVHHP (hydrophobic) YGYGGLYGGLYGLGY (hydrophilic)



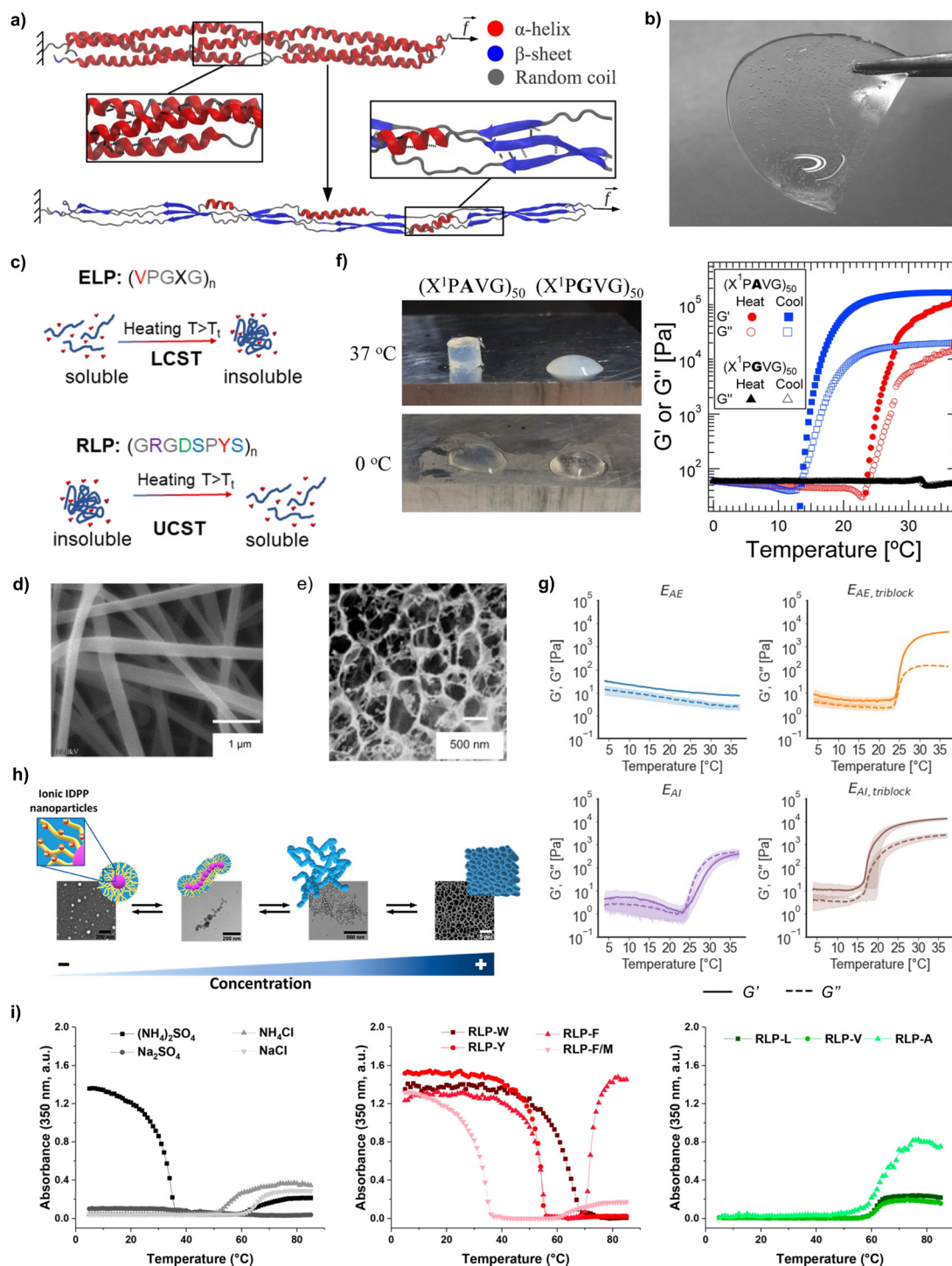


Fig. 2 (a) Helical-to- β -sheet transition in a coiled-coil protein (reproduced and adapted with permission from ref. 26. Copyright 2012, American Chemical Society); (b) hydrogel produced from recombinant collagen-like polypeptides (reproduced and adapted with permission from ref. 28. Copyright 2021, American Chemical Society); (c) thermoreponsiveness of recombinant polypeptides with LCST behaviour (e.g., ELPs) and UCST behaviour (e.g., RLPs) (reproduced and adapted with permission from ref. 33. Copyright 2023, American Chemical Society); (d) fibres produced via electrospinning of aqueous solutions of SELPs (reproduced and adapted with permission from ref. 34. Copyright 2010, American Chemical Society); (e) porous films from BAB copolymeric ELPs after hydration in PBS (reproduced and adapted with permission from ref. 35. Copyright 2005, American Chemical Society); (f) photograph and temperature-dependent rheology ($\omega = 100 \text{ rad s}^{-1}$, $\gamma_0 = 1\%$) of 20% (w/w) solutions ELPs containing plastic blocks (reproduced and adapted with permission from ref. 36. Copyright 2015, American Chemical Society); (g) rheological characterisation (storage modulus G' and loss modulus G'') as a function of triblock or alternate arrangements for neutral and positively charged ELP solutions in Milli-Q water (15 wt%) during a temperature sweep between 4 and 37 °C (reproduced and adapted with permission from ref. 37. Copyright 2022, American Chemical Society); (h) supramolecular assembly of ionic ELPs above T_i as a function of concentration (reproduced and adapted with permission from ref. 38. Copyright 2020, American Chemical Society); (i) UV-vis turbidity measurements of RLPs with different solvents and amino acid substitutions (reproduced and adapted with permission from ref. 39. Copyright 2021, American Chemical Society).



One notable helical structure in Nature is collagen, the most abundant protein in the human body.⁴⁰ Collagen self-assembles into a right-handed triple helix, and its tandem repeat is GXY. In mammalian collagen, proline (P) and 4-hydroxyproline (O) are the most common amino acids in the X and Y positions,⁴¹ respectively. 4-Hydroxyproline is a post-translational modification (PTM) of proline, thus requiring the additional co-expression of prolyl hydroxylases that are absent in bacterial cells.⁴² Thus, many researchers have focused on a simpler variant of collagen, namely unhydroxylated bacterial collagen-like polypeptides (CLPs) consisting of GPX tandem repeats²⁸ (Fig. 2b). The extracellular secretion of CLPs has been demonstrated, which could lead to simple and more cost-effective purification and scale-up.⁴³ The lack of O residues in bacterial CLPs does not impeded their self-assembly into triple helices,⁴⁴ but comes at the expense of a reduced mechanical stability compared to hydroxylated collagen due to the formation of fewer hydrogen bonds between their constituent strands.⁴⁵ Much like human collagen in bones, bacterial CLPs also showed the ability to biomineralise inorganic materials such as hydroxyapatite crystals, silver nanoparticles, and silica nanoparticles.⁴⁶

Another common type of helix in Nature are α -helices in keratin.⁴⁷ The helical structure of keratin is held together by hydrogen bonds between N-H groups and C=O groups, while intermolecular disulfide bonds render it insoluble.⁴⁸ The sequence of keratin contains polar and nonpolar residues in a heptad pattern denoted as *abcdefg*, where *a* and *d* are always nonpolar.⁴⁹ Additionally, keratin is often encountered as a higher-order quaternary structure called a coiled-coil, in which α -helices bundle together to create a stable superhelix⁵⁰ that is the main structural component of hair, nails, skin and wool.⁵¹ The successful expression of recombinant keratin-like proteins has been reported, and their self-assembly into fibrils was confirmed.⁵² When compared against coatings produced from keratin extracted from human hair, recombinant keratin-like proteins led to more uniform coatings, which displayed increased biological activity. This was attributed to the increased homogeneity and the lack of contaminants in recombinant keratins. Moreover, the self-assembly of coiled-coil fibrils into higher order structures, such as hydrogels, can be modulated by pH, ionic strength, temperature, or sequence composition.⁵³ As an example, Dooling and Tirrell⁵⁴ synthesised seven different recombinant proteins with coiled-coil domains that differed by a single amino acid mutation. This allowed them to self-assemble hydrogels with a range of relaxation behaviours, spanning 5 orders of magnitude (from 0.22 ± 0.13 to 1608 ± 135 s).

2.1.2. Intrinsically disordered proteins. Many structural proteins in Nature display a high content of disordered regions that typically arise from a high abundance of either charged and polar residues or secondary structure disrupters such as proline and glycine in their sequence.^{55,56} Although it is generally believed that the 3D conformation of a protein is necessary for its functional role, intrinsically disordered

proteins (IDPs) challenge this by displaying biological activity despite lacking a defined 3D structure under physiological conditions.⁵⁷ Of particular interest within the field of biomaterials are elastomeric proteins, such as elastin and resilin. Elastin is a key protein present in extracellular matrix (ECM),⁵⁸ whereas resilin is found in the cuticle and wings of most insects.⁵⁹ These IDPs have inspired the design of elastin-like polypeptides (ELPs) and resilin-like polypeptides (RLPs). Both ELPs and RLPs display reversible deformation and high resilience, due to their conformationally free chains.⁶⁰ This feature has been leveraged to develop ELP- or RLP-based elastomeric biomaterials in the form of hydrogels,⁶¹ films^{35,62,63} or fibrous meshes.³⁴ Another key feature of ELPs and RLPs is their thermoresponsiveness and the presence of a transition temperature (T_t). Specifically, ELPs have a lower critical solution temperature (LCST), whereas RLPs display an upper critical solution temperature (UCST)³³ (Fig. 2c). This makes ELPs and RLPs promising biopolymers for applications as injectable biomaterials for healthcare applications, such as wound healing,⁶⁴ tissue engineering,⁶⁵ drug delivery,⁶⁶ 3D printing bioinks⁶⁷ or adhesives,⁶⁴ to name a few (Fig. 2d and e). Furthermore, this thermoresponsiveness simplifies the purification of ELPs and RLPs *via* temperature cycling,⁶⁸ avoiding the need for more complex and expensive techniques such as affinity chromatography.

ELP tandem repeats are formed by the pentapeptide VPGXG, where X can be any amino acid excluding proline.⁶⁹ The X amino acid directly impacts the T_t of ELPs, which arises from hydrophobic interactions between ELP chains. Below T_t , ELPs adopt random coil conformations due to the formation of an ordered hydrophobic hydration shell around the hydrophobic residues. Above T_t , the water molecules lose their ordered structure and ELP chains can self-assemble *via* intermolecular hydrophobic interactions.⁷⁰ The value of T_t depends on the hydrophobicity of the amino acid residue in the X position:⁷¹ hydrophilic residues (*e.g.*, glutamic or aspartic acid) increase T_t , whereas hydrophobic residues (*e.g.*, as alanine, tryptophan, or isoleucine) reduce it.⁷¹ For instance, ELPs with a 46% ionic weight fraction displayed a T_t of 32 °C, while ELPs with a 45% non-ionic weight fraction had a T_t of 26 °C.³⁸ The X amino acid also impacts the viscoelasticity of ELP hydrogels: ELP containing VPGIG (hydrophobic) blocks had a storage modulus (G') 4.6× larger, and loss modulus (G'') 29.5× larger than when the block was VPGE (hydrophilic).³⁷ Another common elastin-inspired building block widely used contains an alanine in the third position, and different variations of this tandem repeat have been proposed, such as VPAVG or IPA VG. Incorporating these blocks into the sequence of ELPs led to the formation of stiffer hydrogels with hysteresis between solvation and desolvation.³⁷ To that end, Glassman *et al.*³⁶ showed that 20 wt% solutions of (XPGVG)₅₀ (where X was either 20% valine, 80% isoleucine, or 60% valine, 40% isoleucine) formed a turbid liquid at 37 °C. On the contrary, (XPAVG)₅₀ formed a stiff hydrogel under the same conditions (Fig. 2f).



Additional methods to modulate the mechanical properties of ELP materials have been identified, mainly in studies related to the synthesis of hydrogels. The arrangement of blocks is one example. Hydrophilic–hydrophobic AB diblock ELPs formed micelles,⁷² whereas BAB triblock arrangements formed a hydrogel network.⁷³ Triblock hydrophobic arrangements with IPAVG blocks and isoleucine as guest amino acid formed hydrogel networks with a storage modulus ($G' = 20\,000 \pm 1100$ Pa) one order of magnitude larger than alternate arrangements with the same amino acid content and molecular weight (MW) ($G' = 1300 \pm 400$ Pa)³⁷ (Fig. 2g). Higher protein molar concentration also promoted the formation of stiffer hydrogels for ELPs (Fig. 2h). As an example, the storage modulus of ELP hydrogels was found to increase from 2.9 kPa for a concentration of 2.5 mM, up to 5.8 kPa for a concentration of 2.8 mM.³⁸ Similarly, higher MW also led to stiffer networks: hydrogels prepared from ELPs with 120 tandem repeats had a 5-fold increase in modulus (above 0.5 MPa) compared to sequences with only 70 repeats. The same study also underscored the effect of processing conditions (*e.g.*, the ionic strength of the solvent) on the mechanical properties of ELP materials: hydrogels prepared in sodium phosphate buffer (pH 7.6) had 2- to 7-fold increases in high frequency moduli than those prepared in water.³⁶ Mixing different types of ELPs is another strategy to control the properties of the resulting materials. Ternary mixtures of ELPs formed coacervates with different degrees of miscibility – from homogeneous mixtures to core–shell structures – depending on the guest amino acid (V, A, $V_{50\%}/A_{50\%}$, or $V_{80\%}/A_{20\%}$), the number of tandem repeats (from 40 to 160), or the concentration (volume fraction Φ from 0.001 to 0.1).⁷⁴ The assembly of coacervates from ELP mixtures exhibited also a pathway-dependent assembly, enabling the synthesis of multiple assemblies from the same starting compounds and providing an additional handle to control the properties of ELP-based materials.⁷⁵

The other aforementioned thermoresponsive protein, resilin, is one of the most resilient materials known,⁷⁶ and has thus inspired the development of RLPs for the manufacture of materials with low stiffness, high extensibility and outstanding resilience and fatigue life. RLPs are typically derived from variations of the tandem repeat GGRPSDSYGAPGGGN,⁷⁷ typically by substituting the central tyrosine by amino acids such as phenylalanine, methionine, tryptophane or leucine. Similarly to ELPs, RLPs have also a high content of glycine and proline that promotes disorder and resilience. But conversely to ELPs, RLP's tandem repeat contains more hydrophilic amino acids, which enables a broader diversity of interchain interaction modes than in ELPs, such as electrostatic, hydrophobic, π – π stacking and hydrogen bonding interactions. The T_t and the material properties for RLP materials can be adjusted by tuning the MW, concentration, ionic strength, or amino acid substitutions.^{78–80} Amino acid mutations in RLP blocks affected the gelation propensity of RLP solutions, with

aromatic residues (tryptophan, tyrosine, phenylalanine) promoting π – π or cation– π interactions that facilitated gelation, something that did not happen when aliphatic (leucine, valine, alanine) residues were used^{39,61} (Fig. 2i). To that end, Dzuricky *et al.*⁸⁰ demonstrated that the interaction between cationic arginines and aromatic amino acids were the main drivers of RLP coacervation. Additionally, the T_t of RLPs was observed to increase with increasing MW. RLPs with a MW of 9.5 kDa phase separated at *ca.* 28 °C, whereas RLPs with a MW of 15.8 kDa phase separated at *ca.* 60 °C.⁸¹ The same study also showed that the T_t was lower for dilute RLP solutions.

2.1.3. β -Sheet-rich proteins. β -Sheets are another common 3D conformation found in natural structural proteins. They were first identified in fibroin, a major constituent in silkworm silk,⁸² and have since been detected in other natural materials, such as squid ring teeth (SRT) proteins⁸³ or curli proteins in bacterial biofilms.⁸⁴ These proteins usually self-assemble into β -sheet-rich crystalline regions embedded in a disordered matrix (Fig. 3a). The crystalline regions (with Young's moduli up to 40 GPa (*ref.* 85)) rely on densely-packed hydrogen bonding between aligned strands, and act as strong inter- and intrachain physical crosslinks that mechanically reinforce the material.^{86,87}

The archetypical tandem repeats used to promote the formation of β -sheets are the hexapeptide GAGAGS derived from the silkworm *Bombyx mori*,⁹² or polyalanine-rich polypeptides derived from the major ampullate spidroin of spider silk (such as GAGAAAAAGGAGTS).^{93,94} The degree of crystallinity (and hence, the strength) of silk-based materials directly correlates with the content in GAGAGS or polyalanine regions and with the manufacturing route used (*e.g.*, casting, spinning). Furthermore, the balance between β -sheet-forming blocks and disordered blocks controls the microstructure of these materials: increasing the content of polyalanine blocks in silk-like polypeptides (SLPs) caused a segregation of hydrophobic blocks that changed the morphology of the self-assembled structures formed, from rod-like structures to micelles.⁹⁵ The MW also influenced the self-assembly propensity, with higher MWs leading to the formation of larger aggregates and promoting the transition from disordered to lamellar structures⁸⁸ (Fig. 3b).

SRT proteins (Fig. 3c) also contain a repetitive and alternating crystalline/disordered structure that can form antiparallel β -sheets with turns.⁹⁰ The tandem repeats for SRT are typically formed by PAAASVSTVHHP (the β -sheet-forming block) and YGYGGLYGGYGLGY (the disordered block).⁹⁶ The sequence within the β -sheet-forming domain alternates the position of H and A residues in neighbouring chains, leading to a compact structure, whereas the amorphous blocks show sequence repetition that may contribute to increased mechanical flexibility.⁹⁷ SRTs have been proposed as a promising biopolymer for underwater adhesive applications, due to their high tensile (*ca.* 1.5 MPa) and shear strength (*ca.* 2.5 MPa) in wet conditions.⁸³ The adhesive properties of SRTs were linked to their high content



in β -sheets.⁹⁸ Additionally, the stability of SRTs networks in wet environments was attributed to the high histidine content in their β -sheet-forming blocks,⁸³ because of the versatility of the imidazole side chain in histidines to stabilise protein networks (*e.g.*, *via* hydrogen bonding, metal coordination, or covalent crosslinking).

With respect to SRTs, Vural *et al.*⁹⁹ demonstrated that the energy efficiency of SRT-based actuators with graphene oxide scaled inversely with the number of SRT tandem repeats. This was due to the ability of lower MWs to achieve higher curvatures, whereas SRTs with higher MWs were more restricted in their ability to bend due to the increased

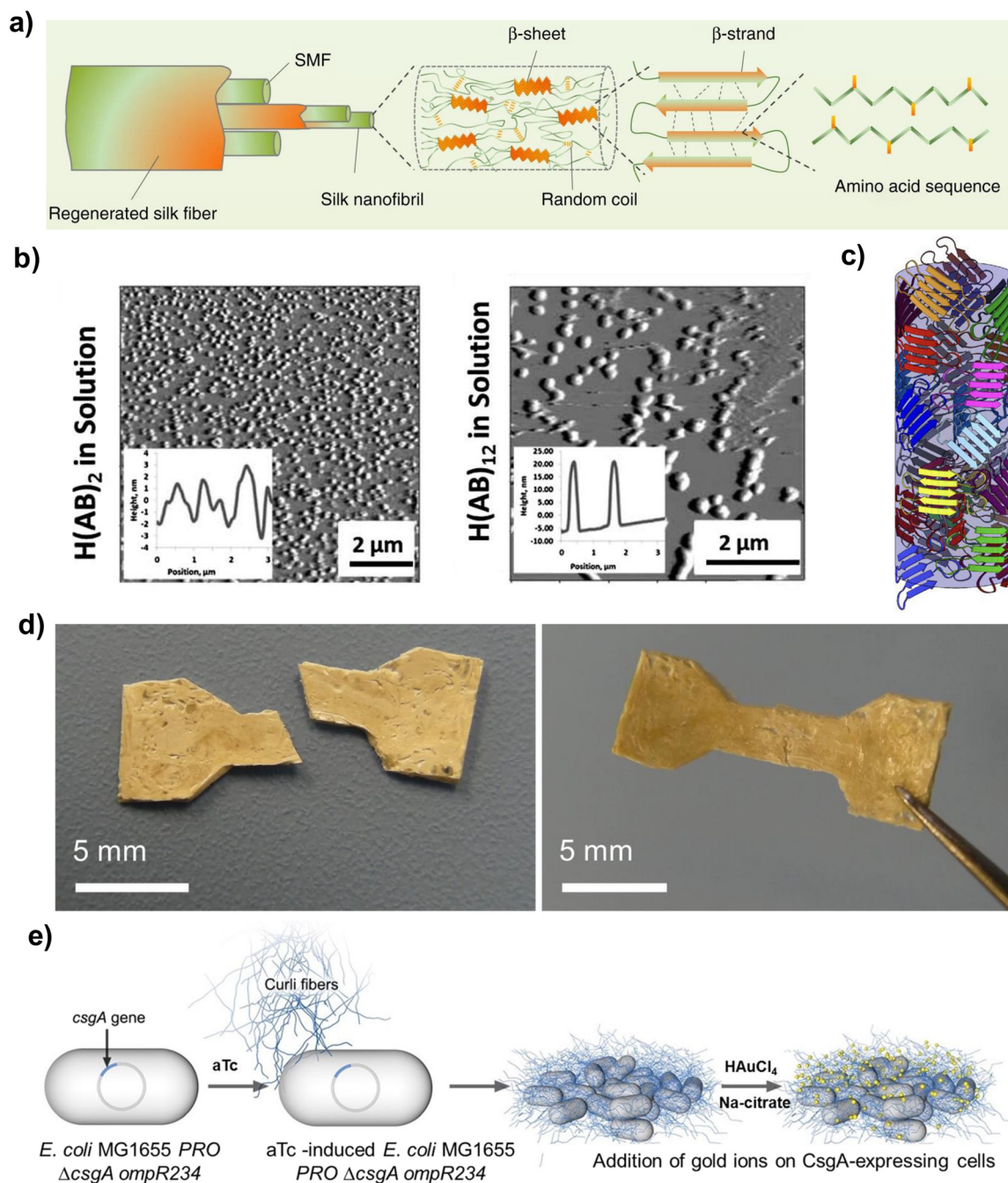


Fig. 3 (a) Schematic representation of the alignment of β -sheet crystals in natural silk fibres (reproduced with permission from ref. 87. Copyright 2017, Nature Portfolio); (b) effect of MW on the self-assembly of silk-like polypeptides, with insets showing the spacing between self-assembled morphologies (reproduced and adapted with permission from ref. 88. Copyright 2014, Elsevier); (c) schematic representation of the isotropic distribution of β -sheet crystals in SRTs (reproduced with permission from ref. 89. Copyright 2017, Nature Portfolio); (d) self-healing properties of RLP-based materials (reproduced with permission from ref. 90. Copyright 2015, Nature Portfolio); (e) schematic representation of the secretion and self-assembly of recombinant curli fibrils that subsequently template the formation of gold nanoparticles (reproduced and adapted with permission from ref. 91. Copyright 2016, American Chemical Society).



physical crosslinks and entanglements. SRTs were also used to develop films with tuneable thermal transport. Below <35% relative humidity, the thermal conductivity of SRT films was independent of the MW. In contrast, when the SRT films were hydrated, the thermal conductivity exhibited an almost linear dependency with the inverse of the number of tandem repeats in the SRT sequence. When analysing the mechanical properties of films, it was reported that an increasing MW did not alter the shear modulus of dry films. In turn, under hydrated conditions, SRTs with higher MW led to films with an increased shear modulus.¹⁰⁰ SRTs have also been used to develop self-propelled microrobots,⁹⁶ which could have applications in minimally invasive medical operations, removal of water contaminants, or targeted cargo delivery.

Curli proteins are another example of structural proteins rich in β -sheets. This protein class is extracellularly secreted and is the main component of bacterial biofilms from *E. coli*. Curli proteins self-assemble into very robust amyloid nanofibres with CsgA as the dominant protein and with a strength comparable to steel.¹⁰¹ The tandem repeat of curli proteins is S(X)₅QXGXGNXA(X)₃Q, and this is repeated five times to make up the amyloid core. S, Q and N residues align in stacks and hydrogen bonds stabilise the structure, which are reinforced by steric zipper interactions between side chain residues of neighbouring chains.^{18,102,103} The tendency of curli proteins to form amyloid fibres enabled the development of the biofilm-integrated nanofiber display (BIND) strategy. This strategy fused CsgA to other functional peptides (e.g., hydroxyapatite binding, gold binding) and used the secretory machinery of *E. coli* to secrete these fusion proteins and produce fibrillar meshes with the ability to i.e., template nanoparticle assembly, provide substrate adhesion, conduct electricity, or covalently immobilise enzymes.^{91,101}

Recombinant proteins inspired by silk, SRT, or curli have been investigated for numerous applications, including tissue regeneration,⁹⁴ engineering fibres,¹⁰⁴ molecular motors⁹⁶ or actuators.¹⁰⁵ The non-covalent nature of β -sheet-rich networks allows them to exhibit fast self-healing kinetics after structural damage^{105–107} (Fig. 3d). It was also shown that this behaviour could be enhanced by applying localised heating to the damage site, due to the thermal mobility of proteins. For instance, damaged SRT-based pneumatic actuators exhibited 2–23 MPa strength after 1 s of healing under heat.¹⁰⁵ In terms of microstructure, SRT-based materials exhibit an anisotropic packing of β -sheet crystals with a well-defined nanoscale size.⁸⁹ In contrast, silk- and curli-like polypeptides tend to self-assemble into fibrillar structures. Additionally, it has been recently demonstrated that SLPs and SRT-based proteins can be thermally moulded from a melt into a solid.^{89,108} This technique is widely used for fossil-based polymers, and thus could facilitate the adoption of recombinant proteins for the manufacture of sustainable and high-performance bioplastics.

2.1.4. Fusion proteins. Fusion proteins combine two or more types of tandem repeats into a single polypeptide chain,

normally with the aim of combining dissimilar mechanical/structural properties into a single biopolymer chain, or to improve specific properties (e.g., strength, toughness). The most widely studied fusion proteins are silk–elastin-like polypeptides (SELPs), in which ordered silk-like β -sheet-forming blocks (GAGAGS) are combined with disordered stimuli-responsive elastin-like blocks (VPGXG) (Fig. 4a). Thus, SELPs bring together the strength and tuneable stiffness of silk and the elasticity and stimuli responsiveness of elastin.^{109,110} These proteins have been used to form hydrogels, films, nanoparticles and fibres with potential applications in fields like drug delivery, injectable hydrogels for tissue engineering, or actuators.^{111–113} SELPs undergo two gelation processes.^{86,114} First, a fast gelation process takes place once T_t is surpassed, due to the fast coacervation of elastin-like blocks. This brings silk-blocks in close proximity, triggering a second, slower gelation process due to the formation of intra-chain β -sheets.¹¹⁵ These β -sheets act as physical crosslinks, increasing in content over time and inducing fibrillar ordering in the materials.¹¹⁶

The properties of SELPs (and other fusion proteins) can be modulated using a variety of strategies. The viscoelastic properties are concentration dependent, with G' and G'' values of SELP increasing in a nonlinear fashion with the wt%.¹¹⁴ The ratio between blocks is another important parameter influencing gelation, with higher silk-to-elastin ratios increasing T_t .^{112,119} However, recent studies hint at the existence of a critical silk-to-elastin ratio above which the silk-like blocks begin to dominate the properties of the material, and the thermoresponsiveness of elastin blocks vanishes, making the gelation quasi-irreversible¹¹⁵ (Fig. 4b). Another relevant parameter for the self-assembly of SELPs is charge of the X amino acid in elastin-like blocks. Like in ELPs, SELPs with uncharged amino acids (e.g., phenylalanine or tyrosine) display a lower T_t than those with charged amino acids (e.g., glutamic acid or lysine).¹¹⁹ Charge also modulates the microstructure of SELP coacervates, promoting microphase separation.¹²⁰ SRT, another β -sheet-forming polypeptide, was also fused to ELPs, thus forming SRT-ELP fusions in which SRT blocks enhance the mechanical properties of the resulting materials in a similar way to silk-like blocks in SELPs.¹²¹ To that end, 12mer, 24mer and 36mer SRT-ELP fusions were reported, ranging from 49.8 to 145.5 kDa. These fusion proteins were used to produce fibres, and the results indicated an enhancement of mechanical properties at higher MWs, with breaking strengths increasing from 303 ± 14 MPa (12mer) to 550 ± 20 MPa (36mer).

SLPs have also been fused to other natural structural proteins, such as mussel foot protein 5 (Mfp5), an IDP used for surface adhesion underwater. When Mfp5 was fused to the end termini of silk blocks, it increased the fibre tensile strength by up to 345% and toughness by up to 1970%.¹²² Golinska *et al.*¹²³ fused SLPs with CLPs with the aim of merging two orthogonal self-assembly mechanisms: the fibrillation ability of SLPs and the tendency to form triple helices of CLPs. They showed that SLP–CLP fusions could



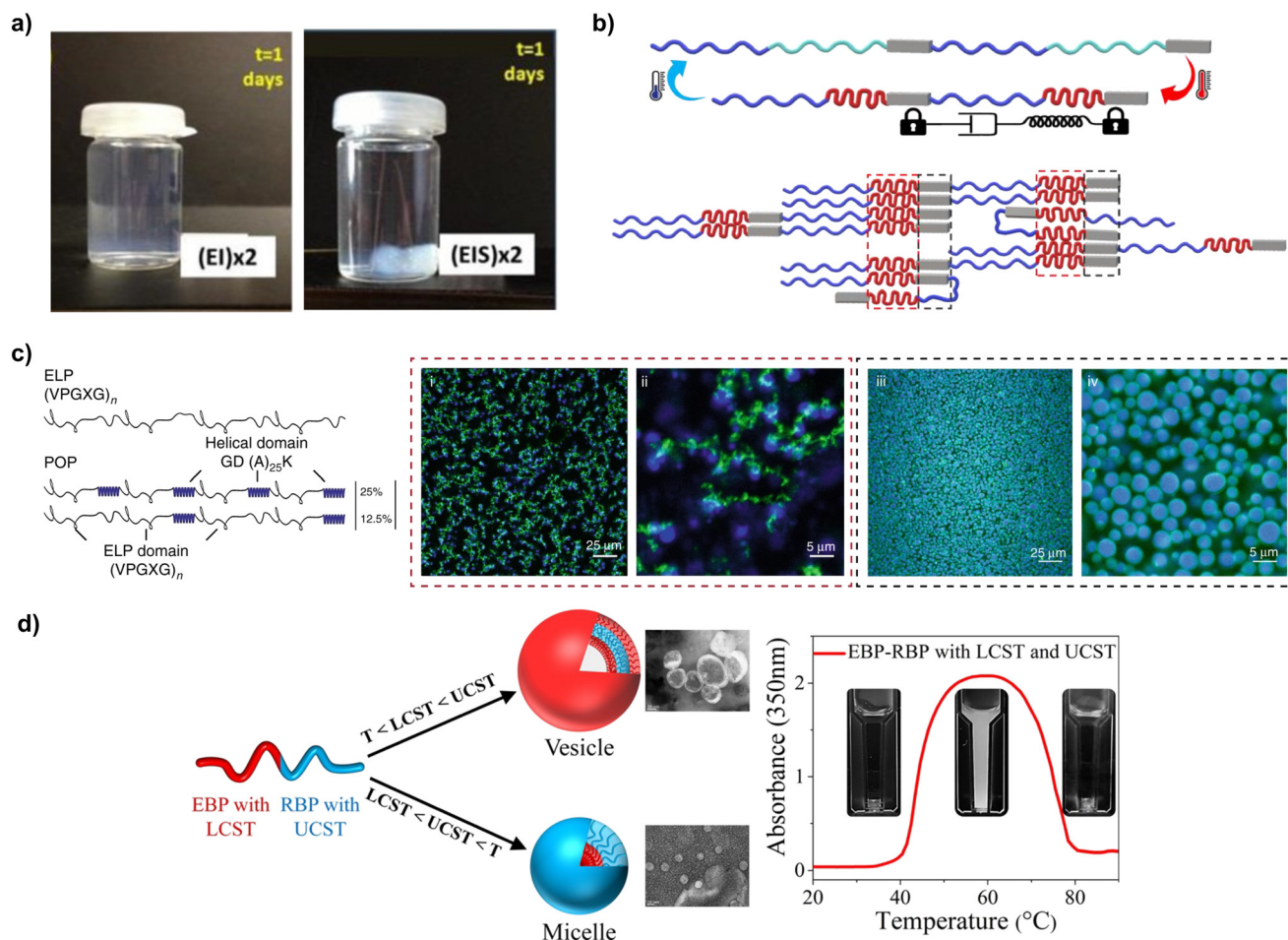


Fig. 4 (a) Adding silk-like blocks to ELPs enhances the robustness and durability of hydrogels produced from those polypeptides (reproduced and adapted with permission from ref. 114. Copyright 2014, American Chemical Society); (b) representation of the elastin-like blocks (red regions) bring silk-like blocks in close proximity (grey regions) facilitating β -sheet formation (reproduced and adapted with permission from ref. 115. Copyright 2022, Elsevier); (c) schematic representation of the sequence and architecture for ELPs and ELP-POP fusions, and the different network architectures that they can form imaged using confocal microscopy (i–iv) (reproduced and adapted with permission from ref. 117. Copyright 2020, Nature Portfolio); (d) ELP-RBP fusions and their dual thermoresponsiveness (reproduced and adapted with permission from ref. 118. Copyright 2021, American Chemical Society).

be applied to synthesise self-healing and pH-responsiveness hydrogels with up to $G' = 1700$ Pa (protein concentration of 2 wt%). In a subsequent study, Rombouts *et al.*²⁷ demonstrated that the self-assembly of CLP trimers was thermoreversible, enabling the synthesis of hydrogels with switchable stiffness controlled by temperature (*ca.* 3 kPa at 40 °C and *ca.* 6 kPa at 20 °C).

ELPs have been fused to other types of blocks too. In a study by Roberts *et al.*, ELP tandem repeats were fused with helical partially ordered polypeptides (POPs), forming porous viscoelastic networks (Fig. 4c). The T_t and microstructure of these networks were controlled by the ratio between ELP and POP regions. Increasing the percentage of POPs from 0–50% (whilst keeping the same ELP block sequence and MW) resulted in a 10 °C difference between solvation and desolvation.¹²⁴ CLP is another helix-forming block that has been fused to ELPs. The tendency of CLP blocks to self-assemble into triple helices caused an increase of the

concentration of ELPs at the edge of the helices formed that CLPs, which lowered their T_t .¹²⁵ However, when the temperature increased beyond the melting temperature of CLPs, the crowding effect disappeared due to the disassembly of the CLP triple helix, leading to a re-dissolution of the fusion proteins.

Materials with dual thermoresponsiveness have also been developed by engineering RLP-ELP fusions. The resulting materials were fully miscible in water below LCST or above UCST but were immiscible between LCST and UCST¹¹⁸ (Fig. 4d). Different nanostructures (namely, disordered, cylindrical, and lamellar) were obtained, with their specific shape determined by temperature, RLP block length or protein concentration. The size of the nanostructures (lamellae, hexagonally packed cylinders, and disordered structures) increased as the RLP fraction increased, or as the concentration of the polymer decreased.¹²⁶ Fusions of three different types of tandem repeats have also been reported.



One example are RLP–ELP–CLP fusions, in which RLP and ELP blocks were used as elastomeric hydrophilic and hydrophobic blocks, respectively. The higher polarity of RLPs aided in the dissolution of RLP–ELP–CLP fusions, whereas the CLP region provided the material with the propensity towards self-assembling into fibrillar structures after incubation in water (0.5 mg mL^{-1}) at 50°C .¹²⁷

2.2. Computer-aided protein design

The 3D conformation of recombinant structural proteins is tightly linked to the mechanical properties of the materials produced from them, but it is difficult to predict that conformation for *de novo* recombinant proteins. Computational modelling can aid in exploring *in silico* the vast design and conformational space for new recombinant protein sequences,¹²⁸ which would reduce the number and cost of time-consuming design–build–test–learn cycles that are used to develop protein-based materials.^{129,130} Furthermore, the effect of manufacturing techniques (*e.g.*, shear, temperature, ionic strength, solvent) can also be incorporated into simulations.¹³¹ There are several computational tools available to that end, depending on the desired level of resolution. At the most fundamental level are quantum mechanical methods, using *e.g.*, density functional theory (DFT).¹³² DFT can aid in assessing molecular geometries, chemical bonding, bandgaps, or cohesive energies, but they are computationally expensive because

they take into account the effects of electrons. Thus, they can only be used to investigate systems below *ca.* 5000 atoms for timescales in the ps range.¹³³ Moving up in length- and timescales, the Rosetta macromolecular modelling tools¹³⁴ enable the prediction of atomistic structures for protein sequences and have been applied to design nanoscale protein blocks with specific 2D or 3D assemblies.¹³⁵

The presence of highly mobile and disordered regions in many natural and recombinant structural proteins makes it also important to assess their dynamic behaviour. To that end, molecular dynamics (MD) simulations are a powerful methodology that can deliver structural, kinetic, and thermodynamic information for recombinant structural proteins. Fully atomistic MD simulations coalesce the influence of electrons with their nuclei and apply the Newtonian equations of motion to study the dynamics of systems of up to millions of atoms and for timescales in the microseconds range. Fully atomistic MD simulations have been successfully applied to many recombinant and natural structural proteins (Fig. 5a), including silk,^{136,137} elastin,^{138,139} collagen,¹⁴⁰ resilin,¹⁴¹ and fusions thereof.^{112,130} These simulations use force fields that accurately calculate the forces and energies for these systems. Some examples of extensively validated force fields for biomolecular simulations include CHARMM,¹⁴² AMBER¹⁴³ or OPLS.¹⁴⁴ However, predicting the secondary structure for *de novo* recombinant proteins *via* fully atomistic MD simulations can be computationally costly, especially if solvent molecules are

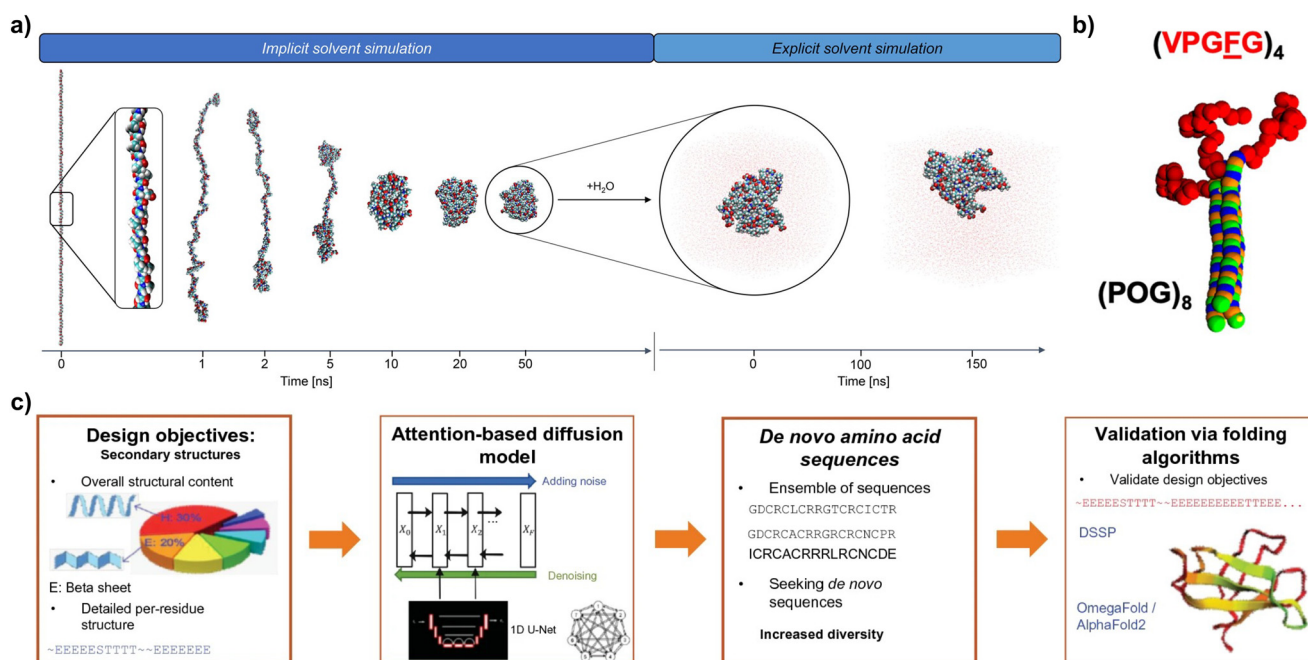


Fig. 5 (a) Screenshots of the folding process (in implicit solvent) and dynamic run (in explicit solvent) of MD simulations for an ELP (reproduced and adapted with permission from ref. 37. Copyright 2022, American Chemical Society); (b) coarse grain model for ELP–CLP conjugates (ELPs in shown red) (reproduced and adapted with permission from ref. 152. Copyright 2019, American Chemical Society); (c) implementation of a workflow for the AI-assisted design *de novo* proteins with target mechanical properties using attention-based diffusion models (reproduced and adapted with permission from ref. 128. Copyright 2023, Springer).



explicitly represented in the simulation. To overcome this issue, several advanced sampling techniques exist, among which replica exchange MD (REMD) simulations are a very common one¹⁴⁵ that has been successfully applied study silk,^{146–148} elastin,¹⁴⁹ collagen¹⁵⁰ or silk–elastin^{112,151} protein-based materials. REMD can accelerate the prediction of the protein secondary structure by focusing on conformational sampling, but at the expense of losing dynamical information. At larger scales, coarse grained (CG) MD simulations can be applied¹⁵² (Fig. 5b). CG MD sacrifices atomistic resolution by lumping several atoms into one single bead, enabling the exploration of 10–100× larger length- and timescales than MD simulations.^{93,151–153}

Finally, ML is emerging as a promising tool for high-throughput discovery of new protein-based materials (Fig. 5c). ML methods can be classified into supervised learning, unsupervised learning, reinforcement learning and semi-supervised learning.¹²⁸ These methods have shown promising results in connecting protein sequences with their secondary structure (forward design), or in predicting amino acid sequences based on secondary structure design objectives (inverse design).^{148,154} ML methodologies are heavily dependent on the availability of reliable and annotated data, for which the establishment of comprehensive and accessible databases for protein-based materials is key.¹⁵⁵

3. Synthetic biology for materials design and fabrication

3.1. DNA assembly methods for genes encoding protein biopolymers

Essential to the biomanufacture of recombinant structural proteins is the strategic design of the DNA encoding for those proteins. The repetitive nature of many structural protein-based materials in nature presents certain challenges for their synthesis using microbial cells, including genetic instability due to increased recombination events (often with a high GC content);¹⁵⁶ difficulties in proper protein expression and folding leading to misfolded proteins and the formation of inclusion bodies; and a significant metabolic burden on the host cells that can hinder cell growth and productivity. These features present significant challenges in assembling short DNA fragments into genes that faithfully encode the desired full-length protein biopolymers.

To express structural proteins in microbial platforms, cloning is typically the first step, but genes encoding for repetitive proteins are unavoidably repetitive. This can make conventional PCR-based cloning methods fail, because non-specific binding can occur during primer annealing in repeated sequences. Other cloning strategies such as restriction-enzyme-based strategies can also be challenging, due to the presence of restriction sites in many repetitive sequences. Nonetheless, the rapid progression in synthetic biology's gene assembly methodologies has paved the way for the fabrication of multifunctional protein-based block

copolymers.¹⁵⁷ The concept of gene assembly allows for the consolidation of various functionalities into a singular protein entity, thereby enabling the stepwise creation of intricate protein substances which can be customised in their composition, quantity, and sequential structure of amino acid blocks. To that end, this section reviews the guiding principles behind several novel DNA assembly methods that have been reported to facilitate the construction of repetitive gene sequences. It also describes the diverse synthetic biology tools available for such gene construction, emphasising their advantages and disadvantages.

3.1.1. Combinatorial codon scrambling. Combinatorial codon scrambling is a method designed to facilitate PCR-based gene synthesis of repetitive proteins by optimising gene sequences to be less repetitive. This approach takes advantage of the redundancy of amino acids' codon representation to find sequences that minimise repetition, thus avoiding the issues that PCR faces with repetitive sequences. Tang and Chilkoti developed a codon-scrambling algorithm that searches for different combinations of synonymous codons for a given repetitive protein fragment and identifies the least repetitive coding sequence.¹⁵⁸ Using this method, they generated a wide range of repeated protein biopolymers, including elastin-like polypeptides, resilin-like polypeptides, and collagen-like polypeptides.

3.1.2. Iterative engineering via recursive directional ligation. Recursive directional ligation (RDL) is a widely used technique for constructing multimeric genes from monomeric DNA, aiming for specific sizes through an orderly process.¹⁵⁹ This method involves iteratively adding oligomers to a linearised vector with a compatible sticky end DNA sequence, through sequential digestion and ligation procedures (Fig. 6a). The use of type IIS restriction enzymes is recommended for the digestion steps to avoid the introduction of unwanted amino acid sequences into the final protein biopolymer structure.¹⁶⁰ This method was successfully implemented to construct a synthetic gene encoding for an ELP containing 108 pentapeptide repeats VPGXG (X = valine, glycine, or alanine).¹⁶¹ In another study, four rounds of RDL were employed to assemble a library of expression plasmids with 16 monomers of the core sequence of dragline silk protein from the species *Araneus ventricosus*, producing proteins with MWs up to 52.6 kDa.¹⁶² Nonetheless, while RDL is a highly precise technique, this precision comes at the cost of labour-intensive cloning cycles required to achieve the target gene dimensions.¹⁵⁹ Moreover, the monomer gene can become circularised, thus preventing its insertion into the vector.

3.1.3. Concatemerisation: engineering multimeric genes through DNA self-ligation. Concatemerisation involves fusing selected monomeric DNA segments that have compatible overhanging sequences, where type IIS restriction sites are the recommended choice.¹⁶³ This process starts with the synthesis of double-stranded DNA monomers, which can be obtained through different methods, such as enzymatic cleavage of plasmid DNA (Fig. 6b). Subsequently, DNA



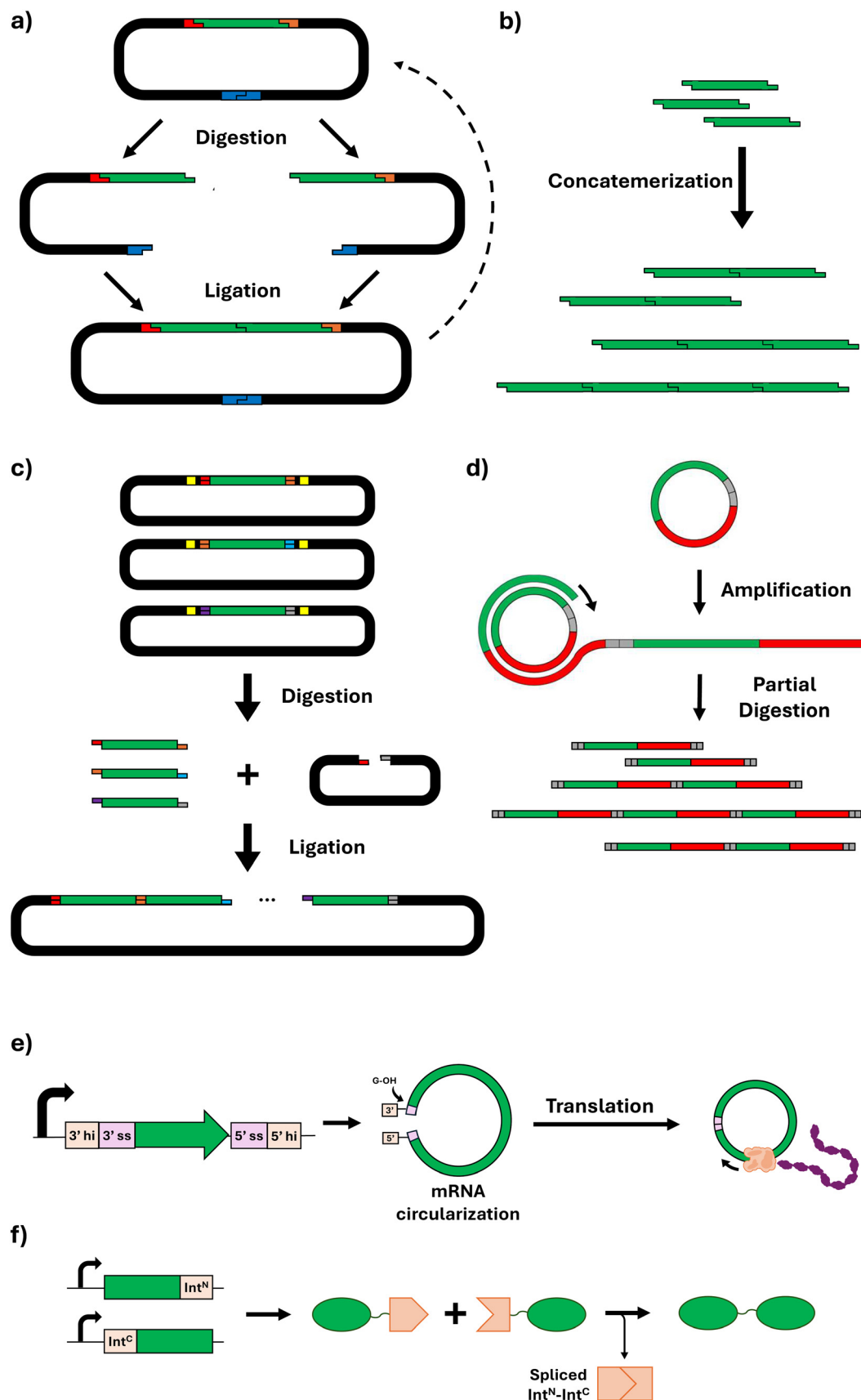


Fig. 6 Schematic representation of DNA assembly methods for genes encoding protein biopolymers. (a) Recursive directional ligation; (b) concatemerization; (c) Golden Gate assembly; (d) overlap extension rolling circle amplification; (e) circular mRNA amplification (hi: half intron; ss: splice site); (f) ligation of protein monomers via split intein system (Int^N : N-intein; Int^C : C-intein).



monomers are linked together to form oligomers of variable lengths using a T4 DNA ligase or by applying thermal cycling methods. Concatemerisation stands out for its simplicity and speed, offering the potential to generate a collection of multimeric genes in a single reaction. Using this methodology, diverse libraries have been developed for the efficient high-throughput production and examination of silk-elastin-like protein biopolymers with a variety of mechanical characteristics and responses to environmental stimuli.¹¹⁹ However, this method presents substantial challenges in precisely engineering the final chain length and sequence composition of multimeric genes, especially when multiple distinct monomers are joined.¹⁶⁴

3.1.4. Precision assembly using the Golden Gate method.

The Golden Gate method is one of the most widely used methods for gene and pathways assembly in synthetic biology. This technique, which allows for the elimination of the enzyme's recognition sites post-digestion by employing type IIS restriction enzymes, facilitates the assembly of multiple DNA fragments with custom overhangs in a single, seamless reaction (Fig. 6c).¹⁶⁵ By designing synthetic DNA segments, or synthons, flanked with Golden Gate-compatible overhangs, it is relatively simple to efficiently construct complex gene circuits^{166,167} and genes with repetitive DNA sequences where precision and fidelity are crucial.^{168,169} It has been effectively applied to assemble genes encoding for structures such as dragline-silk-mimicking proteins, amyloid-spider silk, elastin-like polypeptides, and proteins found in squid ring teeth,^{97,170–172} with lengths varying between 4 and 16 repeat units.

This method typically allows for reliable ligation of up to ten fragments simultaneously; beyond this number, the likelihood of successful assembly tends to decrease. For larger constructs, sequential rounds of the Golden Gate process may be employed, although this approach depends on the availability of distinct overhang sequences for each fragment.¹⁷² Alternatively, larger constructs can be generated by combining Golden Gate with other assembly methods/strategies. For instance, Tang and Chilkoti produced ELPs with 150 repeat units by using Golden Gate to join 30 repeat unit monomers generated by combinatorial codon scrambling.¹⁵⁸ Additionally, due to its combinatorial and modular nature, Golden Gate assembly can also facilitate the production of fusion proteins.^{173,174}

3.1.5. Overlap extension rolling circle amplification.

Overlap extension rolling circle amplification (OERCA) is a unique restriction-free process that amplifies circular DNA molecules to produce long, single-stranded DNA (ssDNA) with repetitive sequences.^{164,175,176} In brief, a circular DNA template is first designed to contain the repetitive sequences that encode the desired protein (Fig. 6d). Upon initiation, the DNA polymerase enzyme starts at a primer and continuously replicates around the circular template multiple times without stopping, generating a long ssDNA with tandem repeats of the encoded protein sequence.¹⁷⁵ This ssDNA can then be transcribed into mRNA, which in turn is translated

into a repetitive protein or polypeptide chain. The repetitive nature of the products allows for the high-yield synthesis of proteins with repeating amino acid sequences. Amiram *et al.* used this technique to generate 2500 bp ELPs from a 90 bp monomer gene encoding 5 repeats of the hexapeptide AVPG VG.¹⁷⁵ This approach was also applied for the construction of genes encoding for squid ring teeth (SRT) proteins, generating a mixture of DNA fragments with 4 to 11 repeats depending on the size of the DNA monomer.⁹⁷

3.1.6. The circular mRNA method. Circular mRNA strategies (Fig. 6e) were developed to counter some of the issues associated with traditional DNA cassette assembly techniques for repetitive structural proteins – such as DNA recombination, gene fragment loss, and halted protein translation, among others. The circular mRNA method uses a circular mRNA encoding a single peptide unit as a limitless template for protein synthesis.¹⁷⁷ Such circular mRNAs are created by bridging the 5' splice site of an exon to the 3' site of the same or subsequent exon, forging a covalent loop.¹⁷⁸ The circular nature of mRNAs allows for an increased stability compared to linear mRNAs due to the absence of exposed 5' and 3' ends, which are typical targets for ribonuclease enzymes.¹⁷⁹ This structural resilience against RNase degradation leads to a prolonged lifespan for circular mRNAs within the cell, enhancing their protein production efficiency. This looped configuration provides ribosomes with a continuous template, enabling the synthesis of protein biopolymers with repeated sequences until the ribosomes detach.¹⁸⁰ Using this method, Liu *et al.* produced the polymers of spider silk protein MaSp1 and flagelliform silk-like protein with MW over 110 and 88 kDa, respectively.¹⁸¹ However, much like concatemerisation, this method does inherently produce a wide range of protein biopolymers with variable monomer counts. Consequently, achieving a protein biopolymer with a precise MW can be challenging, compared to protein formation from linear mRNA templates.¹⁶⁴

3.1.7. Synthesis through ligation of LMW protein monomers. Another useful approach to manufacture high MW (HMW), highly repetitive proteins is their synthesis *via* the ligation/polymerisation of low MW, non-repetitive protein units, herein referred to as protein monomers, through the formation of new peptide bonds between monomers. The ligation of protein monomers can be achieved by different enzymatic and chemical methods. In this section we will focus on intein-mediated ligation for the construction of HMW proteins with material properties.

Inteins are protein segments that can excise themselves from a precursor protein and simultaneously ligate the remaining (extein) sequences with a peptide bond, without the need for external enzymes or energy sources.¹⁸² In this process known as protein splicing, the intein entity is absent from the ultimate protein sequence, thereby enabling a seamless peptide ligation.¹⁸³ Split inteins (SIs), representing a variant where the intein is divided into two segments, have demonstrated significant utility in the assembly of very HMW proteins (Fig. 6f). This is exemplified



in the construction of dragline spider silk proteins, such as major ampullate spidroin 1 (MaSp1). Through the application of SI ligation, two individually expressed segments comprising 96 repeats were successfully ligated, yielding a silk protein fusing 192 repeats with a resultant MW of 556 kDa.¹⁸⁴ Beyond the synthesis of repetitive silk proteins, SI-mediated ligation has been employed in the biosynthesis of high-MW mussel foot proteins (Mfps) from *Mytilus galloprovincialis*.¹⁸⁵ Additionally, further expanding the applications of SIs for protein-based materials, this approach was used to perform multiple consecutive ligation steps within microbial cells, achieving the production of protein biopolymers inspired by titin (a protein conferring muscle tissues with passive strength, damping capacity, and rapid mechanical recovery) with exceptionally high MWs of >5 MDa.¹⁸⁶

3.1.8. Direct synthesis of DNA. The synthetic biology field is undergoing rapid transformation due to the reduction in costs for synthetic DNA synthesis, a change that is streamlining the production of protein biopolymers for various applications, and materials are no exception to that. This is owed to advances in DNA synthesis technology, including enzymatic methods that offer higher purity and longer lengths. As a result, researchers now can directly order the complex custom design genes encoding these polymers, instead of relying on the intricate and sometimes cumbersome process of gene assembly previously described. This streamlined approach saves significant time and resources, bypassing the need for labour-intensive assembly and cloning steps. In addition to this, it allows researchers to make complex custom designs that would be more difficult to achieve otherwise. Therefore, the direct procurement of synthetic genes is rapidly becoming a cost-effective and preferred method for acquiring precise genetic material necessary to produce protein biopolymers, facilitating easier and faster research in this area.¹⁸⁷

3.2. Microbial host systems for protein biopolymer expression

Leveraging living organisms for material manufacturing takes advantage of their innate, evolved cellular mechanisms. The capability for organism engineering is tied to the knowledge of its metabolism and genome, with computational tools predicting genetic elements from genomic data. Optimal engineering frameworks require well-characterised genetic components and established genetic modification techniques. Consequently, foundational synthetic biology for materials has centred on well-understood unicellular organisms such as bacterial hosts *Escherichia coli* or *Bacillus subtilis*, yeast such as *Saccharomyces cerevisiae*, *Komagataella phaffii*, and recently also industrially relevant mammalian cell systems. Such organisms facilitate the swift development of genetic circuits, which must be carefully tailored to the host to prevent interference with cellular functions. Beyond unicellular hosts, there have also been advances in using

complex organisms such as genetically modified plants, insects, and animals for synthesising protein biopolymers,¹⁶⁴ though this is out of scope for this review. The following section primarily explores unicellular hosts used for protein synthesis with materials applications. Their advantages and disadvantages are summarised in Table 2.

3.2.1. Bacterial systems: *Escherichia coli* and *Bacillus subtilis*. *E. coli* has been generally favoured for producing recombinant structural proteins due to their genetic malleability, cost-effectiveness, rapid growth, and scalability.^{188,189} It has also served as a chassis to produce other protein biopolymers, including SLPs, ELPs, RLPs, or CLPs. For instance, researchers explored the expression of bacterial CLPs (without prolyl hydroxylation), such as ScI2 from *Streptococcus pyogenes*.¹⁹⁰ These proteins, compatible with the bacterial translation machinery, were successfully expressed in *E. coli* with yields of 19 g L⁻¹ in a high cell density fed-batch process. Later, Merrett *et al.* developed an alternative CLP-based biomaterial platform from a modified subdomain of ScI2 capable of forming stable, homogeneous hydrogels.²⁸ Bioprocessing efforts have also focused on fusion proteins such as SELPs.¹⁹¹ Using a toxin–antitoxin system, Barroca *et al.*²⁰ were able to achieve yields of 12.8 g L⁻¹ in an antibiotic-free media.

After synthesising protein biopolymers, their distribution within the host organism requires additional considerations. These newly formed biopolymers may be found in different cellular locations such as the cytosol, on the cell surface, or within the extracellular space. Recent observations suggest certain proteins, including SLPs or RLPs can undergo phase separation, leading to the formation of condensates specifically at the bacterial cell poles.¹⁹² Furthermore, it is possible to leverage the ability of *E. coli* to secrete curli proteins, thereby transforming the bacteria into continuous “living” factories.^{193,194}

Meanwhile, the Gram-positive bacterium *B. subtilis* has also been widely used in industrial protein production, given its well characterised genetic background, its amenability to high-density fermentation, and the existence of reliable genetic manipulation technologies.¹⁹⁵ *B. subtilis* is particularly relevant as a chassis due to its high protein-secreting ability.¹⁹⁶ Indeed, Xie *et al.* have engineered *B. subtilis* capable of secreting silk proteins TasA, which spontaneously assemble into fibres on the cell surface.¹⁹⁷ In a different study, TasA fibres were functionalized with a mussel foot protein and an engineered hydrophobin-like protein, thus creating a material with underwater adhesion properties.¹⁹⁸

As previously mentioned, the expression of repetitive and HMW recombinant structural proteins in microbial hosts can bring along some issues. Specifically, aminoacyl-tRNA shortages, for example, glycyl-tRNA during the production of glycine-rich proteins (such as collagen, elastin, silk), can lead to premature stop codons and truncated proteins.¹⁹⁹ Boosting the levels of these tRNAs in host cells can therefore significantly improve protein biopolymer production. As an



Table 2 Comparison of the advantages and disadvantages for different host systems used in the production of protein biopolymers

Host	Advantages	Disadvantages
<i>E. coli</i>	- Genetic malleability - Cost-effectiveness - Rapid growth - Scalability	- Limited secretion ability - Limited PTMs for recombinant eukaryotic protein production - Premature stop codons and truncated proteins due to aminoacyl-tRNA shortages
<i>B. subtilis</i>	- Well-characterised genetic background - Reliable genetic manipulation technologies - Superior secreting ability compared to <i>E. coli</i> - Amenability to high-density fermentation	- Limited PTMs for recombinant eukaryotic protein production - Premature stop codons and truncated proteins due to aminoacyl-tRNA shortages
<i>S. cerevisiae</i>	- Very extensive and standardised synthetic biology tools; established genome engineering tools - Can perform some eukaryotic PTMs - Robustness under fermentation conditions	- PTMs with high immunogenicity - Cannot perform very complex mammalian PTMs
<i>K. phaffii</i>	- Standardised synthetic biology tools - Superior secretion ability relative to <i>S. cerevisiae</i> - Glycosylation patterns similar to mammalian cells - High protein titers - Robustness under fermentation conditions	- Requires methanol induction - Some PTMs differ from higher eukaryotes (e.g. N-glycosylations)
Mammalian cells	- Superior secretion ability - Can perform complex and humanised PTMs	- Lower production levels - High-cost fermentation

example, enhancing the internal pool of tRNAGly through the overexpression of glyVXY genes led to a 10- to 35-fold rise in yield for the silk proteins.²⁰⁰ Similarly, increasing both the tRNAAla and tRNAGly pools and supplementing with alanine and glycine elevated the production of MaSp2 proteins, although an excess of these amino acids impeded cell growth and protein synthesis.^{201,202}

The production of recombinant eukaryotic proteins in microbial hosts present the additional challenge of replicating PTMs in microbial hosts. Nevertheless, researchers have been able to engineer an *E. coli* strain capable of producing a silica-mineralising silaffin R5 peptide, native to diatoms, complete with its necessary PTMs.²⁰³ Their method involved evaluating various PTM enzymes from diverse species to determine their effectiveness on the R5 peptide. At the same time, it should be noted that other researchers have been able to leverage the R5 peptide for silica biomineralisation without the need of PTMs, which poses the question of whether those are needed for the R5 function.^{94,204}

3.2.2. Yeast systems: *Saccharomyces cerevisiae* and *Komagataella phaffii*. Advancements in yeast synthetic biology, including the creation of DNA regulatory elements, standardisation, refined methods for constructing and integrating expression constructs into the genome, and the evolution of gene knockout and knockdown technologies,^{205–208} have contributed to a surge in yeast strains that serve as viable platforms for the industrial-scale production of recombinant protein biopolymers. The most common example is the budding yeast, *S. cerevisiae*, which has been demonstrated to

manufacture recombinant protein biopolymers such as a 94 kDa SLPs with yields up to 450 mg L⁻¹.²⁰⁹

One interesting aspect of yeasts as expression hosts is their secretion capacity to the culture medium. To that end, the methylotrophic yeast *K. phaffii* (*Pichia pastoris*) has emerged as an effective platform, due to its proficient secretion system²¹⁰ that can drastically reduce the number of purification steps needed to isolate the target proteins. *K. phaffii* is also favoured for its ability to grow to high cell densities in cost-effective media. Consequently, *K. phaffii* has been effectively used to biosynthesise and secrete various protein biopolymers, such as spider silk proteins. To that end, Jansson *et al.*²¹¹ reported a comparable total yield of SLPs in *K. phaffii* (extracellular and intracellular) similar to that of *E. coli*. However, amount of secreted protein was about one third of the total protein. *K. phaffii* was also used to produce recombinant human collagen, reaching 2.33 g L⁻¹ in a 5 L bioreactor.²¹²

3.2.3. Mammalian cells. Mammalian cell lines can also serve as another option for unicellular heterologous production of recombinant protein biopolymers. They exhibit a significant advantage in expressing HMW protein biopolymers and in their secretion capabilities, which can potentially lead to higher yields and more streamlined purification procedures. Indeed, a series of spider dragline silk genes, including ADF-3, MaSp1, and MaSp2, have been expressed in bovine mammary epithelial alveolar cells (MAC-T) and baby hamster kidney cells (BHK), yielding proteins with MWs ranging from 60 to 140 kDa.²¹³ Moreover, mammalian cells can perform PTMs not possible in bacterial



hosts, exemplified by the production of human-like collagens or spider silk with hydroxylation of prolyl residues in cell lines like HEK293 and HT1080.²¹⁴ These PTMs can be essential for some mechanical properties, such as the aforementioned bending stiffness of triple-helical collagen fibres. Despite these advantages, the production levels from these cell lines remain low, likely due to plasmid copy number limitations and translational challenges posed by repetitive amino acid sequences.²¹⁵ Furthermore, the high cost of mammalian cell fermentation may constrain their use for large-scale industrial production.

3.3. Engineered living materials

The desirable properties observed in many natural materials (e.g., self-assembled and hierarchically organised structures, dynamic responsiveness, autonomous patterning, or self-repair) are usually achieved thanks to the synergistic action of biopolymeric scaffolds and living cells embedded within them.²¹⁶ Inspired by this, the emergent field of engineered living materials (ELMs) aims at hybridising genetically engineered cells and polymeric scaffolds to manufacture synthetic materials that replicate the sophistication of natural materials.²¹⁷ ELMs blur the boundaries between cells and materials, and have been mostly researched for healthcare applications.²¹⁸ Nonetheless, they are expected to impact other fields as well, such as biosensing,²¹⁹ photosynthetic devices,²²⁰ agriculture,²²¹ or patterned materials.²²² Several technological hurdles need to be overcome before ELMs can become a commercially viable option. These hurdles include improving the lifespan and mechanical properties of ELMs, which are currently relatively weak compared to synthetic engineering materials,^{223,224} or accurately understanding the growth and spatial distribution of cellular populations within ELM scaffolds.²²⁵

Current ELMs are generally made by combining fossil-based polymers (polycaprolactone, polyethylene oxide, polyvinyl alcohol, Pluronic F127, polydimethyl siloxane) or natural biopolymers (e.g., agarose, bacterial cellulose, alginate, gelatin, hyaluronic acid) with genetically engineered cells.^{218,226} Several microbial hosts have also been used²²⁷ including *E. coli* due to its aforementioned desirable characteristics; *B. subtilis* due to its protein secretion possibilities; *K. rhaeticus* because of its ability to secrete bacterial cellulose; the yeast *S. cerevisiae*; microalgae (e.g., *Chlorella reinhardtii*, *Platymonas* sp.); or the fungi mycelium,²²⁸ due to its ability to form bulk composite materials. Furthermore, mixed populations (e.g., *K. rhaeticus* and *S. cerevisiae*) have also been used to assemble ELMs.²²² However, the ultimate goal of the ELMs field is to “grow” materials from simple growth media and an inoculum of engineered cells.¹⁵⁷ To that end, recombinant structural proteins are promising scaffolding materials for that purpose: they offer a biocompatible environment for cells, and cells can be engineered to secrete those proteins as they grow. Some proof-of-principle studies have already been

reported, using fusion recombinant proteins inspired in curli proteins,^{193,229,230} elastin,^{67,231} or mussel foot proteins²³² that were able to self-assemble into soft materials. Additionally, the fusion of recombinant structural proteins with functional blocks has shown to confer scaffolds with properties beyond their structural role, such as the ability to bind to other biopolymers or to biomineralise.^{222,233} Here too, predictive computational tools for sequence-processing-properties in new recombinant structural proteins, as well as scaffold-cell interactions²³⁴ are expected to significantly contribute to the development of the ELMs field.

4. Scale-up and commercialisation

While the production of enzymes and biopharmaceuticals at industrial scale has been demonstrated already for decades, the use of microbial cell factories to produce materials is still lagging. Nonetheless, engineering biology to biofabricate materials has been recognised as a key player in sustainability and circular economy ambitions – some estimates indicate that 60% of the physical inputs to the global economy could be derived from biological systems.²³⁵ This has led to the creation of multiple companies active in the field of biopolymeric materials. The main efforts towards commercialisation have focused on β -sheet-rich polypeptides, including silk-like polypeptides (e.g., AMSilk,²³⁶ Bolt Threads,²³⁷ Spiber²³⁸) or squid-ring proteins SRT polypeptides (Tandem Repeat²³⁹). The company Modern Meadow has also reported the production of recombinant collagen using *K. phaffii*.²⁴⁰ This was facilitated by techniques such as codon optimisation to improve gene expression and the integration of these optimised genes into the yeast genome. Furthermore, the concurrent expression of specific hydroxylase enzymes supported necessary PTMs, increasing the production scale of recombinant collagen.

These companies have demonstrated pilot-scale fermentation and purification, using mostly bacterial (mainly *E. coli*) and yeast (*K. phaffii*) hosts.^{9,239–241} The main areas of activity of these companies are the biomedical (tissue engineering, drug delivery) or personal care (e.g., cosmetics) sectors, but there are application examples also for leather alternatives for products like furniture, bags, and shoes. The main barrier to commercialisation for these materials remains cost.⁹ To that end, several strategies could facilitate the downstream purification processes and hence reduce costs. These include extracellular secretion of the target proteins,¹⁹ or incorporating thermoresponsive ELP or RLP blocks that allow for purification *via* thermal cycling.¹⁸⁷

5. Outlook

Nature utilises structural proteins to manufacture complex materials with properties that we cannot currently recreate in synthetic materials, such as a self-assembled and hierarchically organised structure, dynamic responsiveness, autonomous patterning, and self-healing. Such materials



span a wide range of mechanical and structural properties – from soft to stiff, from porous to densely packed, from dynamic to static. Inspired by this, scientists are increasingly using structural proteins to manufacture materials with applications including food, healthcare, adhesives, energy, or textiles. Furthermore, developments in gene synthesis have enabled us to explore beyond the protein sequences that evolution has selected for and create materials from entirely new protein sequences. However, despite the ample opportunity for protein-based materials, there are several challenges that must be addressed to harness their full potential.

On the materials side, the design freedom for recombinant protein sequences makes it impossible to explore their design space *via* experimentation alone. Furthermore, we are still far from understanding how protein sequences and manufacturing conditions regulate the self-assembly of structural proteins into macroscopic materials, as well as the mechanical and structural properties of those materials. To that end, progress in computational simulation techniques and increased computational power (see section 2.2), as well as novel high-throughput characterisation techniques,^{242,243} will accelerate the exploration of the design space and contribute to uncover sequence–manufacturing–property relationships. High-throughput techniques will also enable the application of machine learning in the field of protein-based materials.²⁴⁴ However, a significant challenge will be the regulatory approval for these materials, especially for biomedical applications or materials containing living (genetically modified) cells, such as ELMs.

On the bioprocessing side, large scale fermentation has been successfully demonstrated for a few protein biopolymers (section 4), but there is still uncertainty about the economic feasibility of these biopolymers, especially for non-biomedical applications. One of the primary hurdles is the high cost of gene synthesis, particularly for large, repetitive protein sequences that are common in materials applications. Advances in synthetic biology and automation will be essential to making these materials economically competitive.²⁴⁵ Another challenge is the optimisation of microbial strains, which often struggle to efficiently produce HMW and complex proteins due to the metabolic burden they impose. Strain engineering efforts are needed to improve the expression and folding of these proteins while ensuring the host organism's viability and productivity.²⁴⁶ In addition, bioreactor design must be optimised to support stable and scalable fermentation processes that maintain product quality, consistency, and yield. Furthermore, downstream processing remains labour-intensive and expensive. This is another bottleneck that must be addressed through the development of more efficient purification methods, such as thermoresponsive blocks or simplified secretion strategies.²⁴⁷

Overall, protein-based materials is an exciting field of research at the intersection between materials engineering and synthetic biology. The on-going scale-up and

commercialisation efforts taking place for several applications, be it biomedicine, textiles, or cosmetics, underscore the promise of these materials to address the urgent unmet need for new sustainable polymers. Nonetheless, the multifaceted challenges describe above demand continuing research efforts and multidisciplinary teams to tackle them and make protein-based materials an economically viable alternative to the well-established and cheaper manufacturing processes for fossil-derived polymers.

Data availability

No primary research results, software or code have been included and no new data were generated or analysed as part of this review.

Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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References

- 1 P. Europe, *Plastics - the Facts*, 2022, <https://plasticseurope.org/knowledge-hub/plastics-the-facts-2022/> (accessed 10th April, 2024).
- 2 W. E. Forum, *The New Plastics Economy: Rethinking the future of plastics*, 2016.
- 3 B. P. Federation, https://www.bpf.co.uk/press/Oil_Consumption.aspx (accessed 10th April, 2024).
- 4 W. W. Y. Lau, Y. Shiran, R. M. Bailey, E. Cook, M. R. Stuchtey, J. Koskella, C. A. Velis, L. Godfrey, J. Boucher, M. B. Murphy, R. C. Thompson, E. Jankowska, A. Castillo Castillo, T. D. Pilditch, B. Dixon, L. Koerselman, E. Kosior, E. Favoino, J. Gutberlet, S. Baulch, M. E. Atreya, D. Fischer, K. K. He, M. M. Petit, U. R. Sumaila, E. Neil, M. V. Bernhofen, K. Lawrence and J. E. Palardy, *Science*, 2020, **369**, 1455–1461.
- 5 B. P. Federation, <https://www.bpf.co.uk/industry/Default.aspx> (accessed 10th April, 2024).
- 6 N. C. Abascal and L. Regan, *Open Biol.*, 2018, **8**, 180113.
- 7 I. R. Campbell, M.-Y. Lin, H. Iyer, M. Parker, J. L. Fredricks, K. Liao, A. M. Jimenez, P. Grandgeorge and E. Roumeli, *Annu. Rev. Mater. Res.*, 2023, **53**, 81–104.
- 8 M. J. Harrington and P. Fratzl, *Prog. Mater. Sci.*, 2021, **120**, 100767.
- 9 A. Miserez, J. Yu and P. Mohammadi, *Chem. Rev.*, 2023, **123**, 2049–2111.
- 10 Y. Cao and B. D. Olsen, *Biomacromolecules*, 2022, **23**, 3286–3295.



- 11 D. Kim, Y. Cao, D. Mariappan, M. S. Bono Jr., A. J. Hart and B. Marelli, *Adv. Funct. Mater.*, 2021, **31**, 2005370.
- 12 A. Lee, A. R. Hudson, D. J. Shiwardski, J. W. Tashman, T. J. Hinton, S. Yerneni, J. M. Bliley, P. G. Campbell and A. W. Feinberg, *Science*, 2019, **365**, 482–487.
- 13 K. A. Burke, D. C. Roberts and D. L. Kaplan, *Biomacromolecules*, 2016, **17**, 237–245.
- 14 X. Jia, C. Wang, V. Ranganathan, B. Napier, C. Yu, Y. Chao, M. Forsyth, F. G. Omenetto, D. R. MacFarlane and G. G. Wallace, *ACS Energy Lett.*, 2017, **2**, 831–836.
- 15 G. Matzeu, L. Mogas-Soldevila, W. Li, A. Naidu, T. H. Turner, R. Gu, P. R. Blumeris, P. Song, D. G. Pascal, G. Guidetti, M. Li and F. G. Omenetto, *Adv. Mater.*, 2020, **32**, 2001258.
- 16 J. Guo, C. Li, S. Ling, W. Huang, Y. Chen and D. L. Kaplan, *Biomaterials*, 2017, **145**, 44–55.
- 17 S. Ling, D. L. Kaplan and M. J. Buehler, *Nat. Rev. Mater.*, 2018, **3**, 18016.
- 18 Y. J. Yang, A. L. Holmberg and B. D. Olsen, *Annu. Rev. Chem. Biomol. Eng.*, 2017, **8**, 549–575.
- 19 M. W. T. Werten, G. Eggink, M. A. Cohen Stuart and F. A. de Wolf, *Biotechnol. Adv.*, 2019, **37**, 642–666.
- 20 M. Barroca, P. Rodrigues, R. Sobral, M. M. R. Costa, S. R. Chaves, R. Machado, M. Casal and T. Collins, *Sci. Rep.*, 2016, **6**, 39329.
- 21 K. Xiao, N. S. Malvankar, C. Shu, E. Martz, D. R. Lovley and X. Sun, *Sci. Rep.*, 2016, **6**, 23385.
- 22 J. Strobl, F. Kozak, M. Kamalov, D. Reichinger, D. Kurzbach and C. F. Becker, *Adv. Mater.*, 2023, **35**, 2207586.
- 23 M. J. Buehler, *J. Mater. Res.*, 2006, **21**, 1947–1961.
- 24 H. S. Rapoport and R. E. Shadwick, *Biomacromolecules*, 2002, **3**, 42–50.
- 25 Z. Qin and M. J. Buehler, *Phys. Rev. Lett.*, 2010, **104**, 198304.
- 26 A. Zhmurov, O. Kononova, R. I. Litvinov, R. I. Dima, V. Barsegov and J. W. Weisel, *J. Am. Chem. Soc.*, 2012, **134**, 20396–20402.
- 27 W. H. Rombouts, D. W. de Kort, T. T. H. Pham, C. P. M. van Mierlo, M. W. T. Werten, F. A. de Wolf and J. van der Gucht, *Biomacromolecules*, 2015, **16**, 2506–2513.
- 28 K. Merrett, F. Wan, C.-J. Lee and J. L. Harden, *ACS Biomater. Sci. Eng.*, 2021, **7**, 1414–1427.
- 29 L. K. Hill, M. Meleties, P. Katyal, X. Xie, E. Delgado-Fukushima, T. Jihad, C.-F. Liu, S. O'Neill, R. S. Tu, P. D. Renfrew, R. Bonneau, Y. Z. Wadghiri and J. K. Montclare, *Biomacromolecules*, 2019, **20**, 3340–3351.
- 30 F. A.-O. X. Wang, O. Gnewou, C. Modlin, L. C. Beltran, C. A.-O. Xu, Z. Su, P. Juneja, G. Grigoryan, E. A.-O. Egelman and V. A.-O. Conticello, *Nat. Commun.*, 2021, **12**, 407.
- 31 J. M. Fletcher, R. L. Harniman, F. R. H. Barnes, A. L. Boyle, A. Collins, J. Mantell, T. H. Sharp, M. Antognozzi, P. J. Booth, N. Linden, M. J. Miles, R. B. Sessions, P. Verkade and D. N. Woolfson, *Science*, 2013, **340**, 595–599.
- 32 A. S. Cristie-David, P. Koldewey, B. A. Meinen, J. C. A. Bardwell and E. N. G. Marsh, *Protein Sci.*, 2018, **27**, 1893–1900.
- 33 R. L. Strader, Y. Shmidov and A. Chilkoti, *Acc. Chem. Res.*, 2024, **57**, 302–311.
- 34 K. Nagapudi, W. T. Brinkman, B. S. Thomas, J. O. Park, M. Srinivasarao, E. Wright, V. P. Conticello and E. L. Chaikof, *Biomaterials*, 2005, **26**, 4695–4706.
- 35 K. Nagapudi, W. T. Brinkman, J. Leisen, B. S. Thomas, E. R. Wright, C. Haller, X. Wu, R. P. Apkarian, V. P. Conticello and E. L. Chaikof, *Macromolecules*, 2005, **38**, 345–354.
- 36 M. J. Glassman and B. D. Olsen, *Biomacromolecules*, 2015, **16**, 3762–3773.
- 37 D. López Barreiro, A. Folch-Fortuny, I. Muntz, J. C. Thies, C. M. J. Sagt and G. H. Koenderink, *Biomacromolecules*, 2022, **24**(1), 489–501.
- 38 S. Acosta, L. Poocha, L. Quintanilla-Sierra and J. C. Rodríguez-Cabello, *Biomacromolecules*, 2021, **22**, 158–170.
- 39 C. Garcia Garcia, S. S. Patkar, N. Jovic, J. Mittal and K. L. Kiick, *ACS Biomater. Sci. Eng.*, 2021, **7**, 4244–4257.
- 40 M. D. Shoulders and R. T. Raines, *Annu. Rev. Biochem.*, 2009, **78**, 929–958.
- 41 A. Khedr, M. A. N. Soliman and M. A. Elsayy, in *Peptide Bionanomaterials: From Design to Application*, ed. M. A. Elsayy, Springer International Publishing, Cham, 2023, pp. 1–52, DOI: [10.1007/978-3-031-29360-3_1](https://doi.org/10.1007/978-3-031-29360-3_1).
- 42 C. Rutschmann, S. Baumann, J. Cabalzar, K. B. Luther and T. Hennet, *Appl. Microbiol. Biotechnol.*, 2014, **98**, 4445–4455.
- 43 Z. Abdali, M. Renner-Rao, A. Chow, A. Cai, M. J. Harrington and N.-M. Dorval Courchesne, *Biomacromolecules*, 2022, **23**, 1557–1568.
- 44 P. J. Kaur, R. Strawn, H. Bai, K. Xu, G. Ordas, H. Matsui and Y. Xu, *J. Biol. Chem.*, 2015, **290**, 9251–9261.
- 45 Y. Qiu, C. Zhai, L. Chen, X. Liu and J. Yeo, *ACS Biomater. Sci. Eng.*, 2023, **9**, 3778–3795.
- 46 Z. Abdali, M. Aminzare, A. Chow and N.-M. Dorval Courchesne, *Biomed. Mater.*, 2023, **18**, 015001.
- 47 L. Pauling, R. B. Corey and H. R. Branson, *Proc. Natl. Acad. Sci. U. S. A.*, 1951, **37**, 205–211.
- 48 H. H. Bragulla and D. G. Homberger, *J. Anat.*, 2009, **214**, 516–559.
- 49 D. N. Woolfson, in *Advances in Protein Chemistry*, Academic Press, 2005, vol. 70, pp. 79–112.
- 50 A. Lupas, *Trends Biochem. Sci.*, 1996, **21**, 375–382.
- 51 J. G. Rouse and M. E. Van Dyke, *Materials*, 2010, **3**, 999–1014.
- 52 R. N. Parker, A. Trent, K. L. Roth Stefaniak, M. E. Van Dyke and T. Z. Grove, *Biomed. Mater.*, 2020, **15**, 065006.
- 53 W. A. Petka, J. L. Harden, K. P. McGrath, D. Wirtz and D. A. Tirrell, *Science*, 1998, **281**, 389–392.
- 54 L. J. Dooling and D. A. Tirrell, *ACS Cent. Sci.*, 2016, **2**, 812–819.
- 55 L. Faltova, A. M. Küffner, M. Hondele, K. Weis and P. Arosio, *ACS Nano*, 2018, **12**, 9991–9999.
- 56 L. Malinowska, S. Kroschwald and S. Alberti, *Biochim. Biophys. Acta, Proteins Proteomics*, 2013, **1834**, 918–931.
- 57 V. N. Uversky, *Front. Phys.*, 2019, **7**, 10.
- 58 X. Jia and K. L. Kiick, *Macromol. Biosci.*, 2009, **9**, 140–156.
- 59 S. Lerch, R. Zuber, N. Gehring, Y. Wang, B. Eckel, K.-D. Klass, F.-O. Lehmann and B. Moussian, *BMC Biol.*, 2020, **18**, 195.



- 60 A. J. Bailey, J. Macmillan, P. R. Shrewry, A. S. Tatham, J. Gosline, M. Lillie, E. Carrington, P. Guerette, C. Ortlepp and K. Savage, *Philos. Trans. R. Soc. London, Ser. B*, 2002, **357**, 121–132.
- 61 S. S. Patkar, C. Garcia Garcia, L. L. Palmese and K. L. Kiick, *Biomacromolecules*, 2023, **24**, 3729–3741.
- 62 X. H. To, N. Pebere, N. Pelaprat, B. Boutevin and Y. Hervaud, *Corros. Sci.*, 1997, **39**, 1925–1934.
- 63 B. Lim, J. Kim, M. S. Desai, W. Wu, I. Chae and S.-W. Lee, *Biomacromolecules*, 2023, **24**, 118–131.
- 64 C. Ma, J. Sun, B. Li, Y. Feng, Y. Sun, L. Xiang, B. Wu, L. Xiao, B. Liu, V. S. Petrovskii, L. Bin, J. Zhang, Z. Wang, H. Li, L. Zhang, J. Li, F. Wang, R. Göstl, I. I. Potemkin, D. Chen, H. Zeng, H. Zhang, K. Liu and A. Herrmann, *Nat. Commun.*, 2021, **12**, 3613.
- 65 I. El Maachi, A. Loewen, S. Acosta, S. Rütten, J. C. Rodríguez-Cabello, S. Jockenhoevel and A. Fernández-Colino, *Adv. Funct. Mater.*, 2024, 2313204.
- 66 R. Herrero-Vanrell, A. C. Rincón, M. Alonso, V. Reboto, I. T. Molina-Martinez and J. C. Rodríguez-Cabello, *J. Controlled Release*, 2005, **102**, 113–122.
- 67 M. Dai, J.-P. Belaïdi, G. Fleury, E. Garanger, M. Rielland, X. Schultze and S. Lecommandoux, *Biomacromolecules*, 2021, **22**, 4956–4966.
- 68 A. K. Varanko, J. C. Su and A. Chilkoti, *Annu. Rev. Biomed. Eng.*, 2020, **22**, 343–369.
- 69 S. Roberts, M. Dzuricky and A. Chilkoti, *FEBS Lett.*, 2015, **589**, 2477–2486.
- 70 D. W. Urry, M. M. Long, B. A. Cox, T. Ohnishi, L. W. Mitchell and M. Jacobs, *Biochim. Biophys. Acta, Protein Struct.*, 1974, **371**, 597–602.
- 71 D. W. Urry, D. C. Gowda, T. M. Parker, C.-H. Luan, M. C. Reid, C. M. Harris, A. Pattanaik and R. D. Harris, *Biopolymers*, 1992, **32**, 1243–1250.
- 72 E. Garanger, S. R. MacEwan, O. Sandre, A. Brûlet, L. Bataille, A. Chilkoti and S. Lecommandoux, *Macromolecules*, 2015, **48**, 6617–6627.
- 73 R. E. Sallach, W. Cui, J. Wen, A. Martinez, V. P. Conticello and E. L. Chaikof, *Biomaterials*, 2009, **30**, 409–422.
- 74 J. R. Simon, N. J. Carroll, M. Rubinstein, A. Chilkoti and G. P. López, *Nat. Chem.*, 2017, **9**, 509–515.
- 75 J. Pille, A. Aloï, D. H. T. Le, I. Vialshin, N. van de Laar, K. Kevenaar, M. Merckx, I. K. Voets and J. C. M. van Hest, *Small*, 2021, **17**, 2007234.
- 76 S. Mayavan, N. K. Dutta, N. R. Choudhury, M. Kim, C. M. Elvin and A. J. Hill, *Biomaterials*, 2011, **32**, 2786–2796.
- 77 C. M. Elvin, A. G. Carr, M. G. Huson, J. M. Maxwell, R. D. Pearson, T. Vuocolo, N. E. Liyou, D. C. C. Wong, D. J. Merritt and N. E. Dixon, *Nature*, 2005, **437**, 999–1002.
- 78 S. S. Patkar, Y. Tang, T. Zhang, A. M. Bisram, J. G. Saven, D. J. Pochan and K. L. Kiick, *Biomacromolecules*, 2024, **25**, 2449–2461.
- 79 R. Balu, N. K. Dutta, A. K. Dutta and N. R. Choudhury, *Nat. Commun.*, 2021, **12**, 149.
- 80 M. Dzuricky, B. A. Rogers, A. Shahid, P. S. Cremer and A. Chilkoti, *Nat. Chem.*, 2020, **12**, 814–825.
- 81 F. G. Quiroz and A. Chilkoti, *Nat. Mater.*, 2015, **14**, 1164–1171.
- 82 L. C. Pauling, R. B. Corey and W. T. Astbury, *Proc. R. Soc. London, Ser. B*, 1997, **141**, 21–33.
- 83 A. Pena-Francesch, B. Akgun, A. Miserez, W. Zhu, H. Gao and M. C. Demirel, *Adv. Funct. Mater.*, 2014, **24**, 6227–6233.
- 84 M. F. B. G. Gebbink, D. Claessen, B. Bouma, L. Dijkhuizen and H. A. B. Wösten, *Nat. Rev. Microbiol.*, 2005, **3**, 333–341.
- 85 Z. Xu and M. J. Buehler, *Phys. Rev. E: Stat., Nonlinear, Soft Matter Phys.*, 2010, **81**, 061910.
- 86 A. Ibáñez-Fonseca, D. Orbanic, F. J. Arias, M. Alonso, D. I. Zeugolis and J. C. Rodríguez-Cabello, *Small*, 2020, **16**, 2001244.
- 87 S. Ling, Z. Qin, C. Li, W. Huang, D. L. Kaplan and M. J. Buehler, *Nat. Commun.*, 2017, **8**, 1387.
- 88 O. S. Tokareva, S. Lin, M. M. Jacobsen, W. Huang, D. Rizzo, D. Li, M. Simon, C. Staii, P. Cebe, J. Y. Wong, M. J. Buehler and D. L. Kaplan, *J. Struct. Biol.*, 2014, **186**, 412–419.
- 89 V. Latza, P. A. Guerette, D. Ding, S. Amini, A. Kumar, I. Schmidt, S. Keating, N. Oxman, J. C. Weaver, P. Fratzl, A. Miserez and A. Masic, *Nat. Commun.*, 2015, **6**, 8313.
- 90 V. Sariola, A. Pena-Francesch, H. Jung, M. Çetinkaya, C. Pacheco, M. Sitti and M. C. Demirel, *Sci. Rep.*, 2015, **5**, 13482.
- 91 U. O. S. Seker, A. Y. Chen, R. J. Citorik and T. K. Lu, *ACS Synth. Biol.*, 2017, **6**, 266–275.
- 92 C.-Z. Zhou, F. Confalonieri, M. Jacquet, R. Perasso, Z.-G. Li and J. Janin, *Proteins: Struct., Funct., Bioinf.*, 2001, **44**, 119–122.
- 93 S. Lin, S. Ryu, O. Tokareva, G. Gronau, M. M. Jacobsen, W. Huang, D. J. Rizzo, D. Li, C. Staii, N. M. Pugno, J. Y. Wong, D. L. Kaplan and M. J. Buehler, *Nat. Commun.*, 2015, **6**, 6892.
- 94 Z. Martín-Moldes, D. Ebrahimi, R. Plowright, N. Dinjaski, C. C. Perry, M. J. Buehler and D. L. Kaplan, *Adv. Funct. Mater.*, 2018, **28**, 1702570.
- 95 O. S. Rabotyagova, P. Cebe and D. L. Kaplan, *Biomacromolecules*, 2009, **10**, 229–236.
- 96 A. Pena-Francesch, J. Giltinan and M. Sitti, *Nat. Commun.*, 2019, **10**, 3188.
- 97 H. Jung, A. Pena-Francesch, A. Saadat, A. Sebastian, D. H. Kim, R. F. Hamilton, I. Albert, B. D. Allen and M. C. Demirel, *Proc. Natl. Acad. Sci. U. S. A.*, 2016, **113**, 6478–6483.
- 98 K. Deepankumar, C. Lim, I. Polte, B. Zappone, C. Labate, M. P. De Santo, H. Mohanram, A. Palaniappan, D. S. Hwang and A. Miserez, *Adv. Funct. Mater.*, 2020, **30**, 1907534.
- 99 M. Vural, Y. Lei, A. Pena-Francesch, H. Jung, B. Allen, M. Terrones and M. C. Demirel, *Carbon*, 2017, **118**, 404–412.
- 100 J. A. Tomko, A. Pena-Francesch, H. Jung, M. Tyagi, B. D. Allen, M. C. Demirel and P. E. Hopkins, *Nat. Nanotechnol.*, 2018, **13**, 959–964.
- 101 P. Q. Nguyen, Z. Botyanszki, P. K. R. Tay and N. S. Joshi, *Nat. Commun.*, 2014, **5**, 4945.
- 102 M. L. Evans and M. R. Chapman, *Biochim. Biophys. Acta, Mol. Cell Res.*, 2014, **1843**, 1551–1558.
- 103 E. P. DeBenedictis, D. Ma and S. Ketten, *RSC Adv.*, 2017, **7**, 48102–48112.



- 104 M. Andersson, Q. Jia, A. Abella, X.-Y. Lee, M. Landreh, P. Purhonen, H. Hebert, M. Tenje, C. V. Robinson, Q. Meng, G. R. Plaza, J. Johansson and A. Rising, *Nat. Chem. Biol.*, 2017, **13**, 262–264.
- 105 A. Pena-Francesch, H. Jung, M. C. Demirel and M. Sitti, *Nat. Mater.*, 2020, **19**, 1230–1235.
- 106 Q. Wang, S. Ling, X. Liang, H. Wang, H. Lu and Y. Zhang, *Adv. Funct. Mater.*, 2019, **29**, 1808695.
- 107 Y. Kikuchi, A. Pena-Francesch, M. Vural and M. C. Demirel, *ACS Appl. Bio Mater.*, 2020, **3**, 2507–2515.
- 108 C. Guo, C. Li, H. V. Vu, P. Hanna, A. Lechtig, Y. Qiu, X. Mu, S. Ling, A. Nazarian, S. J. Lin and D. L. Kaplan, *Nat. Mater.*, 2020, **19**, 102–108.
- 109 A. Tarakanova, W. Huang, Z. Qin, D. L. Kaplan and M. J. Buehler, *ACS Biomater. Sci. Eng.*, 2017, **3**, 2889–2899.
- 110 L. Chambre, Z. Martín-Moldes, R. N. Parker and D. L. Kaplan, *Adv. Drug Delivery Rev.*, 2020, **160**, 186–198.
- 111 S. Salinas-Fernández, M. Santos, M. Alonso, L. Quintanilla and J. C. Rodríguez-Cabello, *Appl. Mater. Today*, 2020, **18**, 100500.
- 112 W. Huang, A. Tarakanova, N. Dinjaski, Q. Wang, X. Xia, Y. Chen, J. Y. Wong, M. J. Buehler and D. L. Kaplan, *Adv. Funct. Mater.*, 2016, **26**, 4113–4123.
- 113 Y. S. Wang, W. W. Huang, Y. Wang, X. Mu, S. J. Ling, H. P. Yu, W. S. Chen, C. C. Guo, M. C. Watson, Y. J. Yu, L. D. Black, M. Li, F. G. Omenetto, C. M. Li and D. L. Kaplan, *Proc. Natl. Acad. Sci. U. S. A.*, 2020, **117**, 14602–14608.
- 114 A. Fernández-Colino, F. J. Arias, M. Alonso and J. C. Rodríguez-Cabello, *Biomacromolecules*, 2014, **15**, 3781–3793.
- 115 C. Gonzalez-Obeso, J. C. Rodríguez-Cabello and D. L. Kaplan, *Acta Biomater.*, 2022, **141**, 14–23.
- 116 M. D. Golinska, T. T. H. Pham, M. W. T. Werten, F. A. de Wolf, M. A. Cohen Stuart and J. van der Gucht, *Biomacromolecules*, 2013, **14**, 48–55.
- 117 S. Roberts, V. Miao, S. Costa, J. Simon, G. Kelly, T. Shah, S. Zauscher and A. Chilkoti, *Nat. Commun.*, 2020, **11**, 1342.
- 118 A. Basheer, S. Shahid, M. J. Kang, J. H. Lee, J. S. Lee and D. W. Lim, *ACS Appl. Mater. Interfaces*, 2021, **13**, 24385–24400.
- 119 Q. Wang, X. Xia, W. Huang, Y. Lin, Q. Xu and D. L. Kaplan, *Adv. Funct. Mater.*, 2014, **24**, 4303–4310.
- 120 C. González-Obeso, M. González-Pérez, J. F. Mano, M. Alonso and J. C. Rodríguez-Cabello, *Small*, 2020, **16**, 2005191.
- 121 Y. Li, J. Li, J. Sun, H. He, B. Li, C. Ma, K. Liu and H. Zhang, *Angew. Chem., Int. Ed.*, 2020, **59**, 8148–8152.
- 122 J. Li, B. Jiang, X. Chang, H. Yu, Y. Han and F. Zhang, *Nat. Commun.*, 2023, **14**, 2127.
- 123 M. D. Golinska, M. K. Włodarczyk-Biegun, M. W. T. Werten, M. A. C. Stuart, F. A. de Wolf and R. de Vries, *Biomacromolecules*, 2014, **15**, 699–706.
- 124 S. Roberts, T. S. Harmon, J. L. Schaal, V. Miao, K. Li, A. Hunt, Y. Wen, T. G. Oas, J. H. Collier, R. V. Pappu and A. Chilkoti, *Nat. Mater.*, 2018, **17**, 1154–1163.
- 125 T. Luo and K. L. Kiick, *J. Am. Chem. Soc.*, 2015, **137**, 15362–15365.
- 126 L. A. Navarro, J. J. Ryan, M. Dzuricky, M. Gradzielski, A. Chilkoti and S. Zauscher, *Biomacromolecules*, 2021, **22**, 3827–3838.
- 127 A. Bracalello, V. Santopietro, M. Vassalli, G. Marletta, R. Del Gaudio, B. Bochicchio and A. Pepe, *Biomacromolecules*, 2011, **12**, 2957–2965.
- 128 S. E. Arevalo and M. J. Buehler, *MRS Bull.*, 2023, **48**, 1140–1153.
- 129 W. Huang, D. Ebrahimi, N. Dinjaski, A. Tarakanova, M. J. Buehler, J. Y. Wong and D. L. Kaplan, *Acc. Chem. Res.*, 2017, **50**, 866–876.
- 130 D. López Barreiro, A. Folch-Fortuny, I. Muntz, J. C. Thies, C. M. J. Sagt and G. H. Koenderink, *Biomacromolecules*, 2023, **24**, 489–501.
- 131 H. Shi, T. Ji, C. Zhai, J. Lu, W. Huang and J. Yeo, *J. Mater. Chem. B*, 2022, **10**, 6133–6142.
- 132 A. Paul and T. Birol, *Annu. Rev. Mater. Res.*, 2019, **49**, 31–52.
- 133 G. Gronau, S. T. Krishnaji, M. E. Kinahan, T. Giesa, J. Y. Wong, D. L. Kaplan and M. J. Buehler, *Biomaterials*, 2012, **33**, 8240–8255.
- 134 A. Leaver-Fay, M. Tyka, S. M. Lewis, O. F. Lange, J. Thompson, R. Jacak, K. W. Kaufman, P. D. Renfrew, C. A. Smith, W. Sheffler, I. W. Davis, S. Cooper, A. Treuille, D. J. Mandell, F. Richter, Y.-E. A. Ban, S. J. Fleishman, J. E. Corn, D. E. Kim, S. Lyskov, M. Berrondo, S. Mentzer, Z. Popović, J. J. Havranek, J. Karanicolas, R. Das, J. Meiler, T. Kortemme, J. J. Gray, B. Kuhlman, D. Baker and P. Bradley, in *Methods in Enzymology*, ed. M. L. Johnson and L. Brand, Academic Press, 2011, vol. 487, pp. 545–574.
- 135 T. F. Huddy, Y. Hsia, R. D. Kibler, J. Xu, N. Bethel, D. Nagarajan, R. Redler, P. J. Y. Leung, C. Weidle, A. Courbet, E. C. Yang, A. K. Bera, N. Coudray, S. J. Calise, F. A. Davila-Hernandez, H. L. Han, K. D. Carr, Z. Li, R. McHugh, G. Reggiano, A. Kang, B. Sankaran, M. S. Dickinson, B. Coventry, T. J. Brunette, Y. Liu, J. Dauparas, A. J. Borst, D. Ekiert, J. M. Kollman, G. Bhabha and D. Baker, *Nature*, 2024, **627**, 898–904.
- 136 S. Keten, Z. Xu, B. Ihle and M. J. Buehler, *Nat. Mater.*, 2010, **9**, 359–367.
- 137 G. Bratzel and M. J. Buehler, *Biopolymers*, 2012, **97**, 408–417.
- 138 S. Rauscher and R. Pomès, *eLife*, 2017, **6**, e26526.
- 139 N. K. Li, S. Roberts, F. G. Quiroz, A. Chilkoti and Y. G. Yingling, *Biomacromolecules*, 2018, **19**, 2496–2505.
- 140 A. Gautieri, S. Vesentini, A. Redaelli and M. J. Buehler, *Nano Lett.*, 2011, **11**, 757–766.
- 141 A. Pizzi, L. Sori, C. Pigliacelli, A. Gautieri, C. Andolina, G. Bergamaschi, A. Gori, P. Panine, A. M. Grande, M. B. Linder, F. Baldelli Bombelli, M. Soncini and P. Metrangolo, *Small*, 2022, **18**, 2200807.
- 142 B. R. Brooks, C. L. Brooks III, A. D. Mackerell Jr., L. Nilsson, R. J. Petrella, B. Roux, Y. Won, G. Archontis, C. Bartels, S. Boresch, A. Caffisch, L. Caves, Q. Cui, A. R. Dinner, M. Feig, S. Fischer, J. Gao, M. Hodoscek, W. Im, K. Kuczera, T. Lazaridis, J. Ma, V. Ovchinnikov, E. Paci, R. W. Pastor, C. B. Post, J. Z. Pu, M. Schaefer, B. Tidor, R. M.



- Venable, H. L. Woodcock, X. Wu, W. Yang, D. M. York and M. Karplus, *J. Comput. Chem.*, 2009, **30**, 1545–1614.
- 143 R. Salomon-Ferrer, D. A. Case and R. C. Walker, *WIREs Comput. Mol. Sci.*, 2013, **3**, 198–210.
- 144 W. L. Jorgensen, D. S. Maxwell and J. Tirado-Rives, *J. Am. Chem. Soc.*, 1996, **118**, 11225–11236.
- 145 Y. Sugita and Y. Okamoto, *Chem. Phys. Lett.*, 1999, **314**, 141–151.
- 146 T. Shu, K. Zheng, Z. Zhang, J. Ren, Z. Wang, Y. Pei, J. Yeo, G. X. Gu and S. Ling, *Biomacromolecules*, 2021, **22**, 1955–1965.
- 147 S. T. Krishnaji, G. Bratzel, M. E. Kinahan, J. A. Kluge, C. Staii, J. Y. Wong, M. J. Buehler and D. L. Kaplan, *Adv. Funct. Mater.*, 2013, **23**, 241–253.
- 148 C.-H. Yu, W. Chen, Y.-H. Chiang, K. Guo, Z. Martin Moldes, D. L. Kaplan and M. J. Buehler, *ACS Biomater. Sci. Eng.*, 2022, **8**, 1156–1165.
- 149 A. Tarakanova, W. Huang, A. S. Weiss, D. L. Kaplan and M. J. Buehler, *Biomaterials*, 2017, **127**, 49–60.
- 150 J. Yeo, Y. Qiu, G. S. Jung, Y.-W. Zhang, M. J. Buehler and D. L. Kaplan, *Biomaterials*, 2020, **240**, 119857.
- 151 J. Yeo, W. Huang, A. Tarakanova, Y.-W. Zhang, D. L. Kaplan and M. J. Buehler, *J. Mater. Chem. B*, 2018, **6**, 3727–3734.
- 152 A. Prhashanna, P. A. Taylor, J. Qin, K. L. Kiick and A. Jayaraman, *Biomacromolecules*, 2019, **20**, 1178–1189.
- 153 J. E. Condon, T. B. Martin and A. Jayaraman, *Soft Matter*, 2017, **13**, 2907–2918.
- 154 B. Ni, D. L. Kaplan and M. J. Buehler, *Chem*, 2023, **9**, 1828–1849.
- 155 C. Zhai, T. Li, H. Shi and J. Yeo, *J. Mater. Chem. B*, 2020, **8**, 6562–6587.
- 156 W. V. Srubar, *Trends Biotechnol.*, 2021, **39**, 574–583.
- 157 P. Q. Nguyen, N.-M. D. Courchesne, A. Duraj-Thatte, P. Praveschotinunt and N. S. Joshi, *Adv. Mater.*, 2018, **30**, 1704847.
- 158 N. C. Tang and A. Chilkoti, *Nat. Mater.*, 2016, **15**, 419–424.
- 159 N. Dinjaski, W. Huang and D. L. Kaplan, in *Peptide Self-Assembly: Methods and Protocols*, ed. B. L. Nilsson and T. M. Doran, Springer New York, New York, NY, 2018, pp. 181–192.
- 160 J. R. McDaniel, J. A. MacKay, F. G. Quiroz and A. Chilkoti, *Biomacromolecules*, 2010, **11**, 944–952.
- 161 S.-H. Cheon, B.-Y. Seo, Y.-J. Lee, D. Sim, S.-B. Lee, P. Guruprasath, T. D. Singh, B.-H. Lee, V. Sarangthem and R.-W. Park, *ACS Biomater. Sci. Eng.*, 2020, **6**, 5024–5031.
- 162 T. Liu, A. Liang, Z. Liang, G. Li and F. Wang, *3 Biotech*, 2018, **8**, 252.
- 163 R. E. Sallach, V. P. Conticello and E. L. Chaikof, *Biotechnol. Prog.*, 2009, **25**, 1810–1818.
- 164 K.-K. Tian, Z.-G. Qian and X.-X. Xia, *Adv. Drug Delivery Rev.*, 2023, **194**, 114728.
- 165 C. Engler, R. Gruetzner, R. Kandzia and S. Marillonnet, *PLoS One*, 2009, **4**, e5553.
- 166 E. Weber, C. Engler, R. Gruetzner, S. Werner and S. Marillonnet, *PLoS One*, 2011, **6**, e16765.
- 167 J. M. Pryor, V. Potapov, K. Bilotti, N. Pokhrel and G. J. S. Lohman, *ACS Synth. Biol.*, 2022, **11**, 2036–2042.
- 168 A. Scior, S. Preissler, M. Koch and E. Deuerling, *BMC Biotechnol.*, 2011, **11**, 87.
- 169 M. T. A. Nguyen, M. V. Gobry, N. Sampedro Vallina, G. Pothoulakis and E. S. Andersen, *ACS Synth. Biol.*, 2024, **13**, 963–968.
- 170 D. Wu, A. Koscić, S. Schneider, R. C. A. Dubini, D. C. Rodriguez Camargo, S. Schneider and P. Rovó, *Biomacromolecules*, 2024, **25**, 1759–1774.
- 171 E. Kim, J. Jeon, Y. Zhu, E. D. Hoppe, Y.-S. Jun, G. M. Genin and F. Zhang, *ACS Appl. Mater. Interfaces*, 2021, **13**, 48457–48468.
- 172 B. Dai, C. J. Sargent, X. Gui, C. Liu and F. Zhang, *Biomacromolecules*, 2019, **20**, 2015–2023.
- 173 X. Yang, J. Wei, Y. Wang, C. Yang, S. Zhao, C. Li, Y. Dong, K. Bai, Y. Li, H. Teng, D. Wang, N. Lyu, J. Li, X. Chang, X. Ning, Q. Ouyang, Y. Zhang and L. Qian, *ACS Synth. Biol.*, 2018, **7**, 2331–2339.
- 174 W. Ott, T. Nicolaus, H. E. Gaub and M. A. Nash, *Biomacromolecules*, 2016, **17**, 1330–1338.
- 175 M. Amiram, F. G. Quiroz, D. J. Callahan and A. Chilkoti, *Nat. Mater.*, 2011, **10**, 141–148.
- 176 A. Fire and S. Q. Xu, *Proc. Natl. Acad. Sci. U. S. A.*, 1995, **92**, 4641–4645.
- 177 S. O. Lee, Q. Xie and S. D. Fried, *ACS Cent. Sci.*, 2021, **7**, 1736–1750.
- 178 N. Eger, L. Schoppe, S. Schuster, U. Laufs and J.-N. Boeckel, in *Circular RNAs: Biogenesis and Functions*, ed. J. Xiao, Springer Singapore, Singapore, 2018, pp. 41–52.
- 179 X. Liu, Y. Zhang, S. Zhou, L. Dain, L. Mei and G. Zhu, *J. Controlled Release*, 2022, **348**, 84–94.
- 180 D.-F. Chen, L.-J. Zhang, K. Tan and Q. Jing, *Biotechnol. Biotechnol. Equip.*, 2018, **32**, 116–123.
- 181 L. Liu, P. Wang, D. Zhao, L. Zhu, J. Tang, W. Leng, J. Su, Y. Liu, C. Bi and X. Zhang, *Appl. Environ. Microbiol.*, 2022, **88**, e00028–22.
- 182 A. Nanda, S. S. Nasker, A. Mehra, S. Panda and S. Nayak, *Microorganisms*, 2020, **8**, 2004.
- 183 F. Pinto, E. L. Thornton and B. Wang, *Nat. Commun.*, 2020, **11**, 1529.
- 184 C. H. Bowen, B. Dai, C. J. Sargent, W. Bai, P. Ladiwala, H. Feng, W. Huang, D. L. Kaplan, J. M. Galazka and F. Zhang, *Biomacromolecules*, 2018, **19**, 3853–3860.
- 185 E. Kim, B. Dai, J. B. Qiao, W. Li, J. D. Fortner and F. Zhang, *ACS Appl. Mater. Interfaces*, 2018, **10**, 43003–43012.
- 186 C. H. Bowen, C. J. Sargent, A. Wang, Y. Zhu, X. Chang, J. Li, X. Mu, J. M. Galazka, Y.-S. Jun, S. Ketten and F. Zhang, *Nat. Commun.*, 2021, **12**, 5182.
- 187 J. C. Rodríguez-Cabello, A. Girotti, A. Ribeiro and F. J. Arias, in *Nanotechnology in Regenerative Medicine: Methods and Protocols*, ed. M. Navarro and J. A. Planell, Humana Press, Totowa, NJ, 2012, pp. 17–38.
- 188 G. Bhattacharyya, P. Oliveira, S. T. Krishnaji, D. Chen, M. Hinman, B. Bell, T. I. Harris, A. Ghazitabatabaei, R. V. Lewis and J. A. Jones, *Protein Expression Purif.*, 2021, **183**, 105839.



- 189 M.-T. Chen, C.-F. Hu, H.-B. Huang, Z.-G. Qian and X.-X. Xia, *Angew. Chem., Int. Ed.*, 2022, **61**, e202214177.
- 190 Y. Y. Peng, L. Howell, V. Stoichevska, J. A. Werkmeister, G. J. Dumsday and J. A. M. Ramshaw, *Microb. Cell Fact.*, 2012, **11**, 146.
- 191 T. Collins, M. Barroca, F. Branca, J. Padrão, R. Machado and M. Casal, *Biomacromolecules*, 2014, **15**, 2701–2708.
- 192 S.-P. Wei, Z.-G. Qian, C.-F. Hu, F. Pan, M.-T. Chen, S. Y. Lee and X.-X. Xia, *Nat. Chem. Biol.*, 2020, **16**, 1143–1148.
- 193 A. Y. Chen, Z. Deng, A. N. Billings, U. O. Seker, M. Y. Lu, R. J. Citorik, B. Zakeri and T. K. Lu, *Nat. Mater.*, 2014, **13**, 515–523.
- 194 Y. Cao, Y. Feng, M. D. Ryser, K. Zhu, G. Herschlag, C. Cao, K. Marusak, S. Zauscher and L. You, *Nat. Biotechnol.*, 2017, **35**, 1087–1093.
- 195 K. Zhang, L. Su and J. Wu, *Annu. Rev. Food Sci. Technol.*, 2020, **11**, 295–318.
- 196 J. van Dijk and M. Hecker, *Microb. Cell Fact.*, 2013, **12**, 3.
- 197 Q. Xie, S. On Lee, N. Vissamsetti, S. Guo, M. E. Johnson and S. D. Fried, *Angew. Chem., Int. Ed.*, 2023, **62**, e202305178.
- 198 C. Zhang, J. Huang, J. Zhang, S. Liu, M. Cui, B. An, X. Wang, J. Pu, T. Zhao, C. Fan, T. K. Lu and C. Zhong, *Mater. Today*, 2019, **28**, 40–48.
- 199 A. Heidebrecht and T. Scheibel, in *Advances in Applied Microbiology*, ed. S. Sariaslani and G. M. Gadd, Academic Press, 2013, vol. 82, pp. 115–153.
- 200 X.-X. Xia, Z.-G. Qian, C. S. Ki, Y. H. Park, D. L. Kaplan and S. Y. Lee, *Proc. Natl. Acad. Sci. U. S. A.*, 2010, **107**, 14059–14063.
- 201 H. Cao, S. Parveen, D. Ding, H. Xu, T. Tan and L. Liu, *Sci. Rep.*, 2017, **7**, 11365.
- 202 Y.-X. Yang, Z.-G. Qian, J.-J. Zhong and X.-X. Xia, *Process Biochem.*, 2016, **51**, 484–490.
- 203 A. K. Wallace, N. Chanut and C. A. Voigt, *Adv. Funct. Mater.*, 2020, **30**, 2000849.
- 204 Z. Martin-Moldes, D. L. Barreiro, M. J. Buehler and D. L. Kaplan, *Acta Biomater.*, 2021, **120**, 203–212.
- 205 K. Malcı, E. Watts, T. M. Roberts, J. Y. Auxillos, B. Nowrouzi, H. O. Boll, C. Z. S. D. Nascimento, A. Andreou, P. Vegh, S. Donovan, R. Frangkoudis, S. Panke, E. Wallace, A. Elfick and L. Rios-Solis, *ACS Synth. Biol.*, 2022, **11**, 2527–2547.
- 206 K. Malcı, L. E. Walls and L. Rios-Solis, *Front. Bioeng. Biotechnol.*, 2020, **8**, 589468.
- 207 J. Beal, A. Goñi-Moreno, C. Myers, A. Hecht, M. D. C. de Vicente, M. Parco, M. Schmidt, K. Timmis, G. Baldwin, S. Friedrichs, P. Freemont, D. Kiga, E. Ordozgoiti, M. Rennig, L. Rios, K. Tanner, V. de Lorenzo and M. Porcar, *EMBO Rep.*, 2020, **21**, e50521.
- 208 K. Malcı, N. Jonguitud-Borrego, H. van der Straten Waillet, U. Puodžiūnaitė, E. J. Johnston, S. J. Rosser and L. Rios-Solis, *ACS Synth. Biol.*, 2022, **11**, 3629–3643.
- 209 K. V. Sidoruk, L. I. Davydova, D. G. Kozlov, D. G. Gubaidullin, A. V. Glazunov, V. G. Bogush and V. G. Debabov, *Appl. Biochem. Microbiol.*, 2015, **51**, 766–773.
- 210 Z. Yang and Z. Zhang, *Biotechnol. Adv.*, 2018, **36**, 182–195.
- 211 R. Jansson, C. H. Lau, T. Ishida, M. Ramström, M. Sandgren and M. Hedhammar, *Biotechnol. J.*, 2016, **11**, 687–699.
- 212 L. Ma, X. Liang, S. Yu and J. Zhou, *Bioresour. Bioprocess.*, 2022, **9**, 119.
- 213 A. Lazaris, S. Arcidiacono, Y. Huang, J.-F. Zhou, F. Duguay, N. Chretien, E. A. Welsh, J. W. Soares and C. N. Karatzas, *Science*, 2002, **295**, 472–476.
- 214 S. Frischholz, F. Beier, I. Girkontaite, K. Wagner, E. Pöschl, J. Turnay, U. Mayer and K. von der Mark, *J. Biol. Chem.*, 1998, **273**, 4547–4555.
- 215 S. Grip, A. Rising, H. Nimmervoll, E. Storckenfeldt, S. J. McQueen-Mason, N. Pouchkina-Stantcheva, F. Vollrath, W. Engström and A. Fernandez-Arias, *Cancer Genomics Proteomics*, 2006, **3**, 83–87.
- 216 F. Burla, Y. Mulla, B. E. Vos, A. Aufderhorst-Roberts and G. H. Koenderink, *Nat. Rev. Phys.*, 2019, **1**, 249–263.
- 217 A. P. Liu, E. A. Appel, P. D. Ashby, B. M. Baker, E. Franco, L. Gu, K. Haynes, N. S. Joshi, A. M. Kloxin, P. H. J. Kouwer, J. Mittal, L. Morsut, V. Noireaux, S. Parekh, R. Schulman, S. K. Y. Tang, M. T. Valentine, S. L. Vega, W. Weber, N. Stephanopoulos and O. Chaudhuri, *Nat. Mater.*, 2022, **21**, 390–397.
- 218 A. Rodrigo-Navarro, S. Sankaran, M. J. Dalby, A. del Campo and M. Salmeron-Sanchez, *Nat. Rev. Mater.*, 2021, **6**, 1175–1190.
- 219 T.-C. Tang, E. Tham, X. Liu, K. Yehl, A. J. Rovner, H. Yuk, C. de la Fuente-Nunez, F. J. Isaacs, X. Zhao and T. K. Lu, *Nat. Chem. Biol.*, 2021, **17**, 724–731.
- 220 S. Balasubramanian, K. Yu, A. S. Meyer, E. Karana and M.-E. Aubin-Tam, *Adv. Funct. Mater.*, 2021, **31**, 2011162.
- 221 A. T. Zvinavashe, E. Lim, H. Sun and B. Marelli, *Proc. Natl. Acad. Sci. U. S. A.*, 2019, **116**, 25555–25561.
- 222 C. Gilbert, T.-C. Tang, W. Ott, B. A. Dorr, W. M. Shaw, G. L. Sun, T. K. Lu and T. Ellis, *Nat. Mater.*, 2021, **20**, 691–700.
- 223 S. Molinari, R. F. Tesoriero and C. M. Ajo-Franklin, *Matter*, 2021, **4**, 3095–3120.
- 224 C. M. Heveran and C. J. Hernandez, *Matter*, 2023, **6**, 3705–3718.
- 225 J.-J. Oh, S. Ammu, V. D. Vriend, R. Kieffer, F. H. Kleiner, S. Balasubramanian, E. Karana, K. Masania and M.-E. Aubin-Tam, *Adv. Mater.*, 2024, **36**, 2305505.
- 226 B. An, Y. Wang, Y. Huang, X. Wang, Y. Liu, D. Xun, G. M. Church, Z. Dai, X. Yi, T.-C. Tang and C. Zhong, *Chem. Rev.*, 2023, **123**, 2349–2419.
- 227 T.-C. Tang, B. An, Y. Huang, S. Vasikaran, Y. Wang, X. Jiang, T. K. Lu and C. Zhong, *Nat. Rev. Mater.*, 2021, **6**, 332–350.
- 228 S. Gantenbein, E. Colucci, J. Käch, E. Trachsel, F. B. Coulter, P. A. Rühs, K. Masania and A. R. Studart, *Nat. Mater.*, 2023, **22**, 128–134.
- 229 J. Huang, S. Liu, C. Zhang, X. Wang, J. Pu, F. Ba, S. Xue, H. Ye, T. Zhao, K. Li, Y. Wang, J. Zhang, L. Wang, C. Fan, T. K. Lu and C. Zhong, *Nat. Chem. Biol.*, 2019, **15**, 34–41.
- 230 A. M. Duraj-Thatte, A. Manjula-Basavanna, J. Rutledge, J. Xia, S. Hassan, A. Sourlis, A. G. Rubio, A. Lesha, M. Zenkl,



- A. Kan, D. A. Weitz, Y. S. Zhang and N. S. Joshi, *Nat. Commun.*, 2021, **12**, 6600.
- 231 S. Molinari, R. F. Tesoriero, D. Li, S. Sridhar, R. Cai, J. Soman, K. R. Ryan, P. D. Ashby and C. M. Ajo-Franklin, *Nat. Commun.*, 2022, **13**, 5544.
- 232 Y. Wang, B. An, B. Xue, J. Pu, X. Zhang, Y. Huang, Y. Yu, Y. Cao and C. Zhong, *Nat. Chem. Biol.*, 2021, **17**, 351–359.
- 233 S.-Y. Kang, A. Pokhrel, S. Bratsch, J. J. Benson, S.-O. Seo, M. B. Quin, A. Aksan and C. Schmidt-Dannert, *Nat. Commun.*, 2021, **12**, 7133.
- 234 H. Zhai and J. Yeo, *ACS Biomater. Sci. Eng.*, 2023, **9**, 269–279.
- 235 *Report on engineering biology: opportunities for the UK economy and national goals*, <https://www.gov.uk/government/publications/advice-on-engineering-biology/report-on-engineering-biology-opportunities-for-the-uk-economy-and-national-goals-html>, (accessed 10th April, 2024).
- 236 AMSilk, <https://www.amsilk.com/>, (accessed 10th April, 2024).
- 237 Bolt Threads, <https://boltthreads.com/>, (accessed 10th April, 2024).
- 238 Spiber, <https://spiber.inc/en/>, (accessed 10th April, 2024).
- 239 Tandem Repeat, <https://tandemrepeat.com/>, (accessed 10th April, 2024).
- 240 K. R.-J. L. Dai and D. T. Williamson, *US Pat.*, 11028146B2, 2021.
- 241 D. N. Breslauer, *ACS Biomater. Sci. Eng.*, 2020, **6**, 5980–5986.
- 242 M. Meleties, D. Britton, P. Katyal, B. Lin, R. L. Martineau, M. K. Gupta and J. K. Montclare, *Macromolecules*, 2022, **55**, 1239–1247.
- 243 F. Burla, T. Sentjabrskaja, G. Pletikapic, J. van Beugen and G. H. Koenderink, *Soft Matter*, 2020, **16**, 1366–1376.
- 244 R. L. Martineau, A. V. Bayles, C.-S. Hung, K. G. Reyes, M. E. Helgeson and M. K. Gupta, *Adv. Biol.*, 2022, **6**, 2101070.
- 245 M. A. Morris, R. A. Bataglioli, D. J. Mai, Y. J. Yang, J. M. Paloni, C. E. Mills, Z. D. Schmitz, E. A. Ding, A. C. Huske and B. D. Olsen, *Mol. Syst. Des. Eng.*, 2023, **8**, 227–239.
- 246 O. Gueta, O. Sheinenzon, R. Azulay, H. Shalit, D. S. Strugach, D. Hadar, S. Gelkop, A. Milo and M. Amiram, *Front. Bioeng. Biotechnol.*, 2022, **10**, 913057.
- 247 J. Su, K. Zhao, Y. Ren, L. Zhao, B. Wei, B. Liu, Y. Zhang, F. Wang, J. Li, Y. Liu, K. Liu and H. Zhang, *Angew. Chem., Int. Ed.*, 2022, **61**, e202117538.

