


Cite this: *Nanoscale*, 2023, **15**, 12648

# Atomic-layered V<sub>2</sub>C MXene containing bismuth elements: 2D/0D and 2D/2D nanoarchitectonics for hydrogen evolution and nitrogen reduction reaction†

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The exploitation of two-dimensional (2D) vanadium carbide (V<sub>2</sub>CT<sub>x</sub>, denoted as V<sub>2</sub>C) in electrocatalytic hydrogen evolution reaction (HER) and nitrogen reduction reaction (NRR) is still in the stage of theoretical study with limited experimental exploration. Here, we present the experimental studies of V<sub>2</sub>C MXene-based materials containing two different bismuth compounds to confirm the possibility of using V<sub>2</sub>C as a potential electrocatalyst for HER and NRR. In this context, for the first time, we employed two different methods to synthesize 2D/0D and 2D/2D nanostructures. The 2D/2D V<sub>2</sub>C/BVO consisted of BiVO<sub>4</sub> (denoted BVO) nanosheets wrapped in layers of V<sub>2</sub>C which were synthesized by a facile hydrothermal method, whereas the 2D/0D V<sub>2</sub>C/Bi consisted of spherical particles of Bi (Bi NPs) anchored on V<sub>2</sub>C MXenes using the solid-state annealing method. The resultant V<sub>2</sub>C/BVO catalyst was proven to be beneficial for HER in 0.5 M H<sub>2</sub>SO<sub>4</sub> compared to pristine V<sub>2</sub>C. We demonstrated that the 2D/2D V<sub>2</sub>C/BVO structure can favor the higher specific surface area, exposure of more accessible catalytic active sites, and promote electron transfer which can be responsible for optimizing the HER activity. Moreover, V<sub>2</sub>C/BVO has superior stability in an acidic environment. Whilst we observed that the 2D/0D V<sub>2</sub>C/Bi could be highly efficient for electrocatalytic NRR purposes. Our results show that the ammonia (NH<sub>3</sub>) production and faradaic efficiency (FE) of V<sub>2</sub>C/Bi can reach 88.6 μg h<sup>−1</sup> cm<sup>−2</sup> and 8% at −0.5 V vs. RHE, respectively. Also V<sub>2</sub>C/Bi exhibited excellent long-term stability. These achievements present a high performance in terms of the highest generated NH<sub>3</sub> compared to recent investigations of MXenes-based electrocatalysts. Such excellent NRR of V<sub>2</sub>C/Bi activity can be attributed to the effective suppression of HER which is the main competitive reaction of the NRR.

Received 12th March 2023,

Accepted 10th June 2023

DOI: 10.1039/d3nr01144e

rsc.li/nanoscale

## 1. Introduction

Currently, fossil fuels supply about 80% of the world's energy consumption.<sup>1</sup> However, fossil fuel stocks are limited, and their intensive use will aggravate the problems of greenhouse gas emissions. Therefore, there is an urgent need to increase the utilization of renewable energy resources to meet our ever-increasing energy demands. Hydrogen gas (H<sub>2</sub>), and ammonia (NH<sub>3</sub>) are among the most produced and widely used chemicals in the world.<sup>2,3</sup> Therefore, the production of NH<sub>3</sub> and H<sub>2</sub> through an economic and sustainable process is highly sought

after. The electrochemical hydrogen evolution reaction (HER), and nitrogen reduction reaction (NRR) have attracted widespread attention owing to fossil-fuel-free routes for H<sub>2</sub> and NH<sub>3</sub> synthesis, respectively. To date, precious metals are the most outstanding electrocatalysts for electrochemical H<sub>2</sub> and NH<sub>3</sub> production, especially platinum-based nanomaterials for HER,<sup>4</sup> and palladium-based nanomaterials for NRR.<sup>5</sup> However, the high cost of those precious metals is restricting their practical applications. Therefore, it is urgent to find economical and effective NRR and HER electrocatalysts. Two-dimensional (2D) transition-metal carbides, nitrides, and carbonitrides (MXenes) have received great interest since the discovery of the first MXene (Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>).<sup>6</sup> MXenes usually have a hexagonal close-packed structure with a chemical formula of M<sub>n+1</sub>X<sub>n</sub> (M = transition metal element, X = C, N, n = 1–3). Commonly, MXenes are obtained by selectively etching the “A” elements from their corresponding bulk MAX phases using chemicals, such as

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† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d3nr01144e>



hydrofluoric acid, ammonium bifluoride, or an acidic solution of fluoride salts. Thus, their surfaces are inevitably terminated by a series of functional groups ( $-\text{OH}$ ,  $-\text{F}$ ,  $-\text{O}$ ).<sup>6</sup> Owing to the existence of surface terminations, most MXenes exhibit both metallic conductivity and hydrophilicity,<sup>7</sup> which makes them promising candidates for various applications, such as energy storage and conversion, especially in electrocatalysis.<sup>8,9</sup> Unfortunately, the NRR and HER performances with the most developed MXenes electrocatalysts are still far from satisfactory and not comparable to those of precious metals. Most of the pristine MXenes suffer from large overpotential and low stability in aqueous electrolytes which seriously hinder their large-scale practical applications.<sup>10,11</sup> Additionally, MXenes often exhibit low ammonia yield and unsatisfactory faradaic efficiency (FE),<sup>12,13</sup> because they are usually terminated with inactive functional groups ( $-\text{F}$ ,  $-\text{OH}$ ), which hinder the active metal sites for  $\text{N}_2$  binding.<sup>14,15</sup> Therefore, many attempts have been made to improve the HER and NRR activities of MXenes. Firstly, by introducing or modifying chemical functional groups on its surface.<sup>16–18</sup> It was proved that the electrocatalytic performances of MXenes are significantly influenced by the functional groups on the basal plane ( $-\text{O}$ ,  $-\text{OH}$ ,  $-\text{F}$ ). For instance, MXenes terminated with O species presented relatively low hydrogen adsorption free energy ( $\Delta G_{\text{H}^+}$ ) and would be promising candidates for HER.<sup>11</sup> Secondly, constructing composites of MXenes with other active components (e.g., nanoparticles, oxides, single atoms, and other 2D materials) is another efficient strategy to boost the NRR and HER performance of such 2D layered structures.<sup>19–22</sup> In fact, the MXene-based composites could enhance the catalytic activity by increasing specific surface area (SSA). Among MXene-based composites, 2D/2D heterostructures have been identified as the best choice to ensure the largest contact area and promote electron transfer for improving electrocatalysis reactions.<sup>23</sup>

Recently, Bi-based catalysts have been considered potential NRR candidates owing to their intrinsic catalytic activity, earth-abundance, low cost, and environmentally benign semi-metal.<sup>24</sup> The density functional theory (DFT) calculation also has proved that in Bi-based catalysts, the density state near the Fermi level is mainly localized around the Bi atoms, indicating the ability of Bi to serve as active sites to donate p-electrons and thus activate  $\text{N}_2$ .<sup>25</sup> Currently, promising advances have been achieved for the NRR on Bi-based electrocatalysts including Bi nanosheets,<sup>24</sup> 2D mosaic Bi nanosheets,<sup>25</sup>  $\text{PdBi}_2$  nanoflakes,<sup>26</sup> and bismuthene nanostructures.<sup>27</sup> However, the NRR performance of the reported Bi-based materials is still limited due to their poor electrical conductivity.<sup>25</sup> Thus, combining Bi with conductive nanomaterials is considered a promising strategy to solve this issue. Very recently, Liu *et al.*,<sup>28</sup> successfully anchored metallic Bi on the 2D  $\text{Ti}_3\text{C}_2\text{T}_x$  MXene via a simple liquid phase reduction method. The as-prepared  $\text{Bi}@\text{Ti}_3\text{C}_2\text{T}_x$  composite electrocatalyst achieved an excellent  $\text{NH}_3$  yield of  $28.3 \mu\text{g h}^{-1} \text{cm}^{-2}$  and a large FE of 27.2% at 0.4 V vs. RHE. Their report shows a significant performance for NRR which is

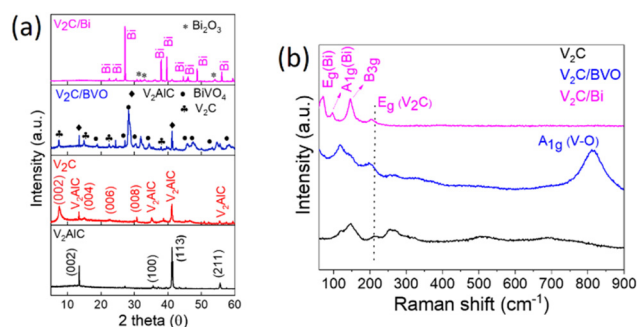
higher than most of the reported Bi-based catalysts. On the other hand, Bi-based catalysts possess poor activity in HER due to the unfavorable free energy of hydrogen adsorption ( $\Delta G_{\text{H}^+}$ ).<sup>29</sup> Thus, a lot of studies have been applied to explore the (photo) catalytic activity of  $\text{H}_2$  production for Bi-based materials. Recently, by adopting different strategies,  $\text{BiOX}$  ( $\text{X} = \text{Cl}$ ,  $\text{Br}$ , and  $\text{I}$ ),<sup>30</sup>  $\text{Bi}_2\text{O}_3$ ,<sup>31</sup> and  $\text{BiVO}_4$ <sup>32</sup> showed extraordinary HER performance. Among Bi compounds,  $\text{BiVO}_4$  has emerged as the most promising candidate for photocatalytic hydrogen production owing to good chemical stability, nontoxicity, and low cost.<sup>32</sup>

In this work, for the first time, we successfully synthesized two different MXene-based nanocomposites including  $\text{V}_2\text{C}/\text{BVO}$  and  $\text{V}_2\text{C}/\text{Bi}$  NPs.  $\text{V}_2\text{C}$  MXene was selected because of its high electrical conductivity compared to the other reported MXenes.<sup>6</sup> Our findings demonstrate that the first nanocomposite ( $\text{V}_2\text{C}/\text{BVO}$ ) consists of  $\text{BiVO}_4$  nanosheets embedded in the few layers of  $\text{V}_2\text{C}$  nanosheets, and the second nanocomposite ( $\text{V}_2\text{C}/\text{Bi}$ ) consists of metallic Bi NPs anchored on the surface of  $\text{V}_2\text{C}$ . The  $\text{V}_2\text{C}$  MXene was obtained through the etching of  $\text{V}_2\text{AlC}$  by  $\text{HCl}/\text{NaF}$ . The as-obtained nanocomposites were directly used as multifunctional electrodes for HER and NRR in acidic and in basic electrolytes, respectively. 2D/2D  $\text{V}_2\text{C}/\text{BVO}$  showed a lower overpotential (384 mV) in 0.5 M  $\text{H}_2\text{SO}_4$  aqueous electrolyte compared to pristine  $\text{V}_2\text{C}$  (823 mV) when applied for HER. Also,  $\text{V}_2\text{C}/\text{BVO}$  proved excellent stability for HER application. Meantime,  $\text{V}_2\text{C}/\text{Bi}$  has been proven to be highly efficient NRR electrocatalyst achieving a record high-yield rate of  $\text{NH}_3$  production with  $88.6 \mu\text{g h}^{-1} \text{cm}^{-2}$  and FE of 8% at  $-0.5 \text{ V}$  vs. RHE. These results indicate that the decoration of a suitable matrix with Bi atoms is beneficial to improve nitrogen fixation performance. To the best of our knowledge, the HER and NRR electrocatalysis performances of the  $\text{V}_2\text{C}/\text{Bi}$  and  $\text{V}_2\text{C}/\text{BVO}$  have not been studied before.

## 2. Results and discussion

### 2.1. Characterization of the catalysts

The crystalline structure of  $\text{V}_2\text{AlC}$ ,  $\text{V}_2\text{C}$ , and composite samples was examined by XRD as shown in Fig. 1a. Firstly, it should be



**Fig. 1** (a) XRD patterns of  $\text{V}_2\text{AlC}$ ,  $\text{V}_2\text{C}$ ,  $\text{V}_2\text{C}/\text{BVO}$ , and  $\text{V}_2\text{C}/\text{Bi}$ . (b) Raman spectra of  $\text{V}_2\text{C}$ ,  $\text{V}_2\text{C}/\text{BVO}$ , and  $\text{V}_2\text{C}/\text{Bi}$ .



mentioned that the as-prepared  $V_2AlC$  precursor presents sharp and strong diffraction peaks at  $2\theta = 13.5^\circ$  and  $41.2^\circ$ , indicating its high crystallinity and purity (PDF file 29-0101).<sup>33</sup> After etching with HCl/NaF solution, several peaks were detected at  $7.5^\circ$ ,  $15^\circ$ ,  $22.6^\circ$ , and  $30.7^\circ$  corresponding to the crystal indexes (002), (004), (006), and (008) of hexagonal  $V_2C$  MXene with a multilayered structure.<sup>33</sup> The presence of small peaks observed at  $13.5^\circ$  and  $41.2^\circ$  suggests the existence of a small amount of unreacted  $V_2AlC$ , indicating the incomplete etching of  $V_2C$ . This observation is consistent with previous studies.<sup>34,35</sup> The same peaks were also detected in  $V_2C/BVO$ , composite confirming the unaltered hexagonal crystalline phase of  $V_2C$  in the  $V_2C/BVO$ . Additional diffraction peaks at  $2\theta = 18.6^\circ$ ,  $28.6^\circ$ ,  $30.7^\circ$ ,  $34.3^\circ$ ,  $39.9^\circ$ ,  $42.5^\circ$ ,  $47.2^\circ$  correspond to the (011), (121), (040), (200), (211), (051) and (042), and (310), respectively, confirming the formation of monoclinic  $BiVO_4$  (JCPDS no. 14-0688).<sup>36</sup> As well, the XRD pattern of  $V_2C/Bi$  shows that all the diffraction peaks can be indexed to Bi rhombohedral crystal structure NPs (code: 01-085-1330).<sup>37</sup> At the same time, the peaks originating from  $V_2AlC$  and  $V_2C$  MXene completely vanished. This can be attributed to the presence of high-density Bi nanoparticles that are fully occupied on the surface of the  $V_2C$ . As a result, it becomes difficult to detect the  $V_2AlC$  and  $V_2C$  peaks. This phenomenon has been observed in previous reports of MXene-based nanocomposites, where the (002) peak of MXenes disappeared after being combined with other components.<sup>38–40</sup> The loss of the (002) peak of  $V_2C$  was attributed to the relatively poor crystallinity of  $V_2CT_x$  and the low signal-to-noise ratio.<sup>39</sup> Low-intensity peaks of  $Bi_2O_3$  can also be observed corresponding to surface oxidation.

The as-prepared samples were further studied by Raman spectroscopy (Fig. 1b). For  $V_2C$ , several bands appeared at approximately 147, 211, 262, 513, and  $691\text{ cm}^{-1}$ . According to the theoretical study of Champagne *et al.*,<sup>41</sup> the peak at  $211\text{ cm}^{-1}$  corresponds to the  $E_g$  mode of  $V_2C(OH)_2$ , the peak at  $262\text{ cm}^{-1}$  corresponds to the  $E_g$  mode of  $V_2C$ , and two weaker peaks at 513 and  $691\text{ cm}^{-1}$  can be assigned the  $A_{1g}$  mode of  $V_2CF_2$  and  $V_2CO(OH)$ , respectively. The  $E_g$  mode is due to the in-plane vibration of the V atoms while the  $A_{1g}$  mode is due to the out-of-plane vibration of the V atoms. As well, for  $V_2C/BVO$  the typical Raman bands of  $V_2C$  are observed with slight blue-shifting, which confirms the interface interaction between  $BiVO_4$  and  $V_2C$ . Other peaks at around 814, 326, and  $118\text{ cm}^{-1}$  were also detected. The most intense Raman peak at about  $814\text{ cm}^{-1}$  is assigned to the symmetric V–O stretching vibrations ( $A_{1g}$ ), the sharp band at  $118\text{ cm}^{-1}$  due to external modes rotation/translation, whereas the weak band is attributed to asymmetric bending vibration ( $B_g$ ) of the  $VO_4^{3-}$  group in  $BiVO_4$ .<sup>42,43</sup> The above results further prove the successful synthesis of the  $V_2C/BVO$  composite. For the second  $V_2C/Bi$  composite, three obvious peaks at 71, 97, and  $146\text{ cm}^{-1}$  are observed. The modes located at 71 and  $97\text{ cm}^{-1}$  can be assigned to the first-order  $E_g$  and  $A_{1g}$  stretching modes, respectively, of the Bi–Bi vibrational modes of rhombohedral Bi NPs.<sup>27,44</sup> The peak at  $206\text{ cm}^{-1}$  is assigned to  $E_g$  of  $V_2C$

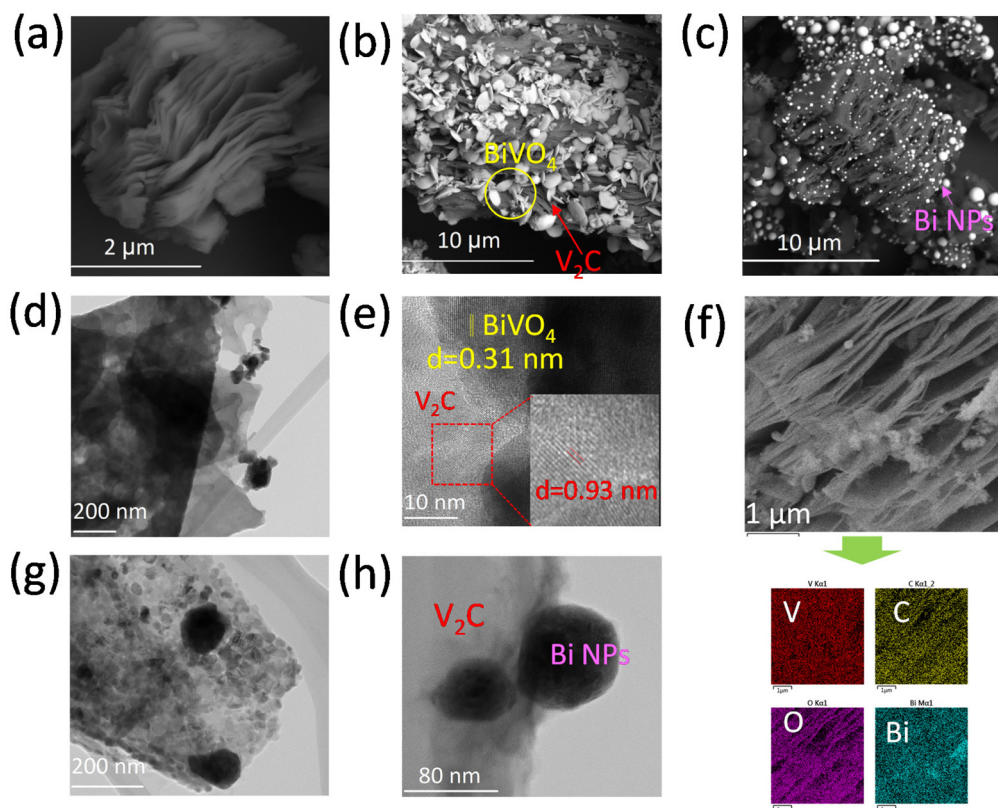
$(OH)_2$  with slight blue-shift from that of pure  $V_2C$  reveals the interaction between Bi NPs and  $V_2C$ , and the predominant band at  $145\text{ cm}^{-1}$  is attributed to the skeleton bent vibrational mode ( $B_{3g}$ ), characteristic of the orthorhombic phase of  $V_2O_5$ ,<sup>45</sup> implying the partial oxidation of the  $V_2C$  MXene. The obtained XRD and Raman results confirm the synthesis of a new composite  $V_2C/Bi$  through the reduction of  $Bi^{5+}$  in  $NaBiO_3$  to metallic Bi NPs. MXene was used as a reducing agent to donate electrons and convert  $NaBiO_3$  to Bi NPs. Previous research has shown that the surface termination groups of MXenes, such as  $-O$ ,  $-OH$ , and  $-F$ , provide excellent adsorption and reduction properties.<sup>46</sup> The reduction of  $Bi^{5+}$  to  $Bi^0$  during the synthesis of  $V_2C/Bi$  composites occurred through a reduction process involving oxidation states of +5, +3, and 0. The presence of  $Bi_2O_3$  trace in the XRD spectrum of the  $V_2C/Bi$  composite confirmed the reduction of  $Bi^{5+}$  to  $Bi^{3+}$ . Consequently, Bi NPs are homogeneously distributed on the surface of  $V_2C$  nanosheets, which are partially oxidized during the synthesis reaction.

The investigation of the  $V_2C$  and  $V_2C/Bi$  elemental composition by EDS is displayed in Fig. S1†. The presence of V and C confirms the successful synthesis of MXene, and the existence of O, F, Cl, and Na further proves the presence of active groups on the surface of MXene. In addition, the peak corresponding to Bi in composites demonstrates the successful introduction of Bi into  $V_2C$  and the formation of a hybrid. More information about the morphology and crystal structure of the material structure was evaluated by SEM, TEM, HRTEM, and selected-area electron diffraction (SAED).

As shown in Fig. 2a, the SEM of  $V_2C$  shows the typical accordion-like multilayer nanostructure. The high-resolution TEM (HRTEM) image in Fig. S2a† reveals the crystalline lattice of  $V_2C$  MXene with an interplanar distance of 0.93 nm, corresponding to its (002) planes, indicating the well-crystallized feature of the as-synthesized  $V_2C$  MXene. The SAED pattern (inset of Fig. S2a†) reveals a hexagonal structure for  $V_2C$  which is in good agreement with the structure determined from the XRD result. As well, for  $V_2C/BVO$ , it can be observed that  $BiVO_4$  flakes were uniformly wrapped on the sheet of  $V_2C$  (Fig. 2b). The TEM image in Fig. 2d further suggests that the  $V_2C/BVO$  is constituted of 2D ultrathin nanosheets. A closer observation (HRTEM in Fig. 2e) revealed the presence of intimate contact and a distinguished contrast between  $BiVO_4$  and  $V_2C$ . Additionally, the interplanar distance of 0.31 and 0.93 nm can be assigned to the (121)  $BiVO_4$ ,<sup>47</sup> and (002) planes of  $V_2C$ , respectively. This observation further corroborates that the as-prepared  $V_2C/BVO$  is a 2D/2D composite with intimate coupling. This junction can increase the contact area between the two materials which can enhance the electron transfer, and as a result, improve the catalytic activity.<sup>32</sup> In addition, the elemental mapping of  $V_2C/BVO$  (Fig. 2f) reveals the existence of homogeneous distribution of C, V, O, and Bi elements.

The selected area diffraction (SAED) image in Fig. S2b† confirms the preservation of the original hexagonal structure of MXene. For  $V_2C/Bi$ , SEM, and TEM/HRTEM images (Fig. 2c, g, and h) reveal the layered structure of  $V_2C$  was well maintained





**Fig. 2** SEM of (a)  $V_2C$ , (b)  $V_2C/BVO$  and (c)  $V_2C/Bi$ , (d) TEM of  $V_2C/BVO$ , (e) HRTEM of  $V_2C/BVO$ , (f) the main elemental components mapping of V, C, O and Bi of  $V_2C/BVO$ , (g) TEM of  $V_2C/Bi$  and (h) HRTEM of  $V_2C/Bi$ .

and Bi nanoparticles were loaded on the surface of the  $V_2C$  MXene. The elemental mapping results in Fig. S2c† further present the uniform distribution of V, C, Bi, and O elements, suggesting that Bi NPs are well dispersed on  $V_2C$ .

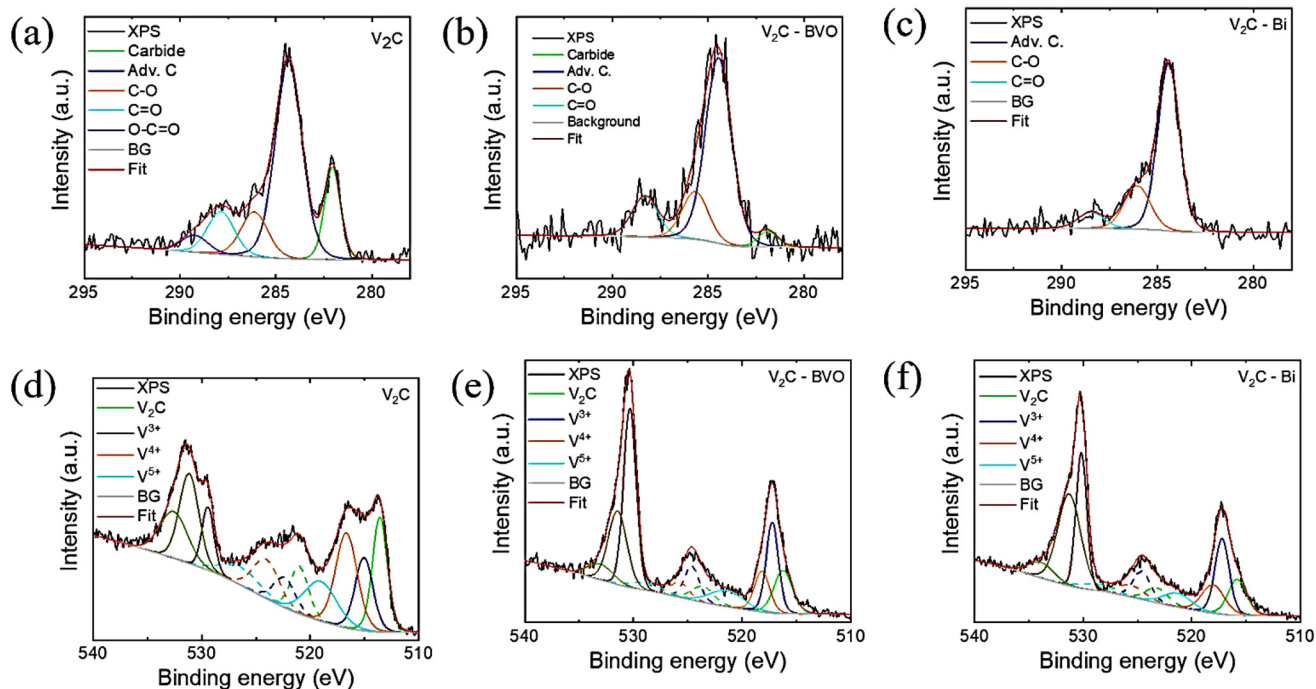
To examine the surface chemical state, the XPS of the samples was measured. Individual bonding states can exhibit slight variability in their position due to an inhomogeneous surface charging. Fig. S3a† shows the XPS survey spectra which confirmed the presence of O, V, F, and C in all three samples. Compared to the original  $V_2C$ , the decrease of the F 1s peak and enhancement of the O 1s peak of composites may be due to the substitution of  $-F$  groups by  $-OH$  groups. The detection of the Bi 4f peak confirms the construction of  $V_2C/BVO$  and  $V_2C/Bi$  heterostructures. The XPS spectrum of C 1s in  $V_2C$  (Fig. 3a) shows the presence of the main peak of C–V (282 eV) belonging to  $V_2C$ .<sup>48</sup> The peaks of adventitious carbon (284.4 eV), C–O (285.9 eV), C=O (287.6 eV), and O–C=O (289 eV) originate from adsorbed species on the surface of the material and from functional groups on the surface of MXene.<sup>38,49–51</sup> For  $V_2C/BVO$ , the peak from carbide V–C (282.6 eV) was still detected (Fig. 3b), indicating the presence of  $V_2C$  on the surface of the material. However, in the case of  $V_2C/Bi$ , the C–V bond peak disappeared (Fig. 3c), implying that the surface of MXene was highly covered with metallic bismuth. The V 2p spectrum (Fig. 3d) shows the characteristic peaks of  $V^{2+}$  species in  $V_2C$  at 513.5

( $2p_{3/2}$ ) and 521.4 ( $2p_{1/2}$ ) eV, as well as  $V^{4+}$  species in  $V_2C$  at 516.5 ( $2p_{3/2}$ ) and 524.2 ( $2p_{1/2}$ ) eV.<sup>48</sup> The obvious peak ( $V^{2+}$ ) confirms the strong metal carbide interaction (V–C) in  $V_2C$ , while peaks of  $V^{4+}$  confirm the existence of vanadium oxide (V–O) on the surface of  $V_2C$ . Simultaneously, the XPS spectrum of V 2p in  $V_2C/BVO$  and  $V_2C/Bi$  composites is shown in Fig. 3e and f. For both composites, the peak related to  $V^{2+}$  species can still be observed but is greatly decreased. Meanwhile, the peak intensity of V–O species increased, which proved the partial oxidation of the surface V atoms during the composite's preparations. Bi 4f peaks (Fig. S3b†) are deconvoluted into two pairs of peaks at 159 eV (Bi 4f<sub>7/2</sub>) and 164 eV (Bi 4f<sub>5/2</sub>) corresponding to the binding energy of  $Bi^{3+}$  in  $V_2C/BVO$ .<sup>32</sup> Similarly for  $V_2C/Bi$  composite, the Bi 4f peaks correspond to oxidized surface ( $Bi^{3+}$ ) of the nanoparticles.<sup>48</sup> The fitting results of the respective elements are listed in Table S1.†

Since the SSA of the catalyst plays a significant role in catalytic activity, we next measured the Brunauer–Emmett–Teller (BET) SSA of the samples by nitrogen absorption and desorption test. As depicted in Fig. S4,† the composite  $V_2C/BVO$  possesses a larger SSA ( $8.56 \text{ m}^2 \text{ g}^{-1}$ ) and pore volume value ( $0.121 \text{ cm}^3 \text{ g}^{-1}$ ) compared to  $V_2C/Bi$  and pristine  $V_2C$ . The larger pore volume and SSA confirmed that the 2D/2D composite with 2D/2D structure possesses an increased number of active sites for redox reactions.





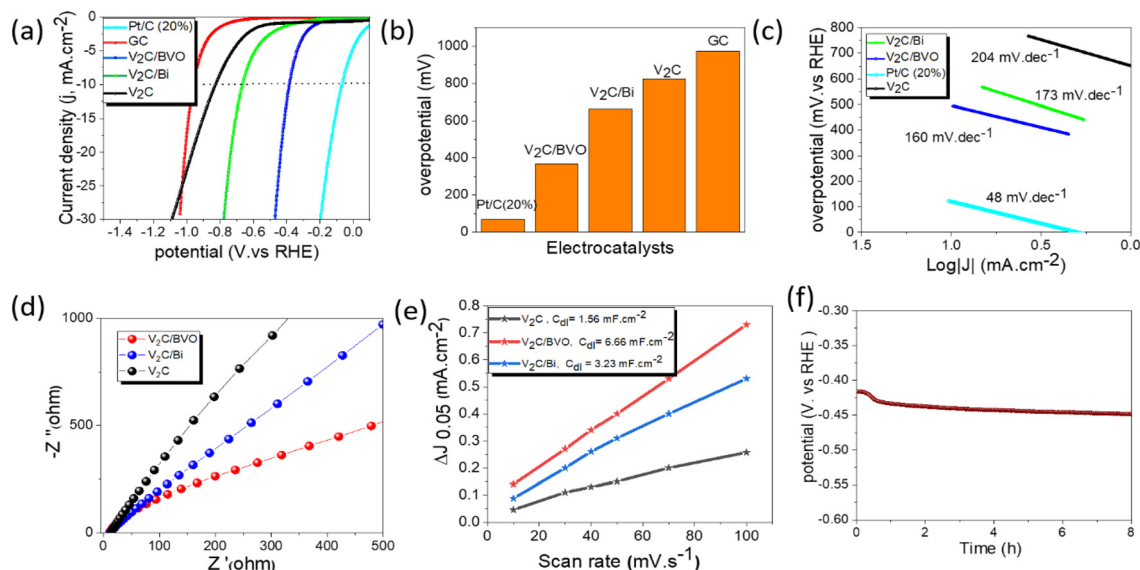


**Fig. 3** High-resolution XPS spectra of C 1s in (a)  $V_2C$ , (b)  $V_2C/BVO$ , and (c)  $V_2C/Bi$ ; high-resolution XPS spectra of V 2p in (d)  $V_2C$ , (e)  $V_2C/BVO$ , and (f)  $V_2C/Bi$ .

### 3.2. Catalytic HER activity

The HER activity of all the catalysts was measured by linear sweep voltammetry (LSV) in 0.5 M  $H_2SO_4$  using three-electrode cells. The HER activity was compared to commercial Pt/C (20 wt%) and glassy carbon electrode (GC). As shown in

Fig. 4a, the GC electrode displayed negligible HER activity which indicates that the observed activity is mainly due to the catalysts. As expected, the Pt/C (20 wt%) shows the highest catalytic activity with an overpotential of 68 mV at  $10 \text{ mA cm}^{-2}$  current density (Fig. 4b). In comparison, the  $V_2C/BVO$  (with an overpotential of 384 mV at  $10 \text{ mA cm}^{-2}$ ) exhibited a much



**Fig. 4** Electrochemical measurements for hydrogen evolution in 0.5 M  $H_2SO_4$  acidic medium: (a) LSV curves, (b) onset overpotential at  $j = 10 \text{ mA cm}^{-2}$  of different electrocatalysts, (c) the corresponding Tafel plots, (d) EIS Nyquist plot of different catalysts, (e) capacitive  $\Delta j$  as a function of scan rates for different samples, and (f) the chronoamperometric measurement for  $V_2C/BVO$  at a constant current density of  $-10 \text{ mA cm}^{-2}$ .



better HER performance than V<sub>2</sub>C/Bi (663 mV) and V<sub>2</sub>C (823 mV), indicating that the formed 2D/2D V<sub>2</sub>C/BVO hybrid provided the active species for the HER. The overpotential ( $\eta_{10}$ ) of the V<sub>2</sub>C/BVO electrocatalyst corresponding to a current density of 10 mA cm<sup>-2</sup> is further compared to these non-noble metal electrocatalysts in Table S2.† It should be noted that the V<sub>2</sub>C/BVO catalyst needs lower overpotential compared to previously reported catalysts. Following this, we evaluated the Tafel slope which is often used to characterize the rate-limiting step of electrocatalytic reaction. The HER mechanism in acidic media is based on two possible reaction pathways namely: (a) Volmer–Heyrovsky and (b) Volmer–Tafel. HER starts with the Volmer reaction, where the hydronium protons acquire electrons on the surface of the electrode and generate an adsorbed hydrogen atom H<sub>ads</sub> as an intermediate state ( $\text{H}_3\text{O}^+ + \text{e}^- \rightarrow \text{H}_{\text{ads}} + \text{H}_2\text{O}$ ). Then, H<sub>2</sub> molecules are dominantly generated through one of two different reaction processes depending on the surface coverage of H<sub>ads</sub>. If the coverage of H<sub>ads</sub> is high, two H<sub>ads</sub> combine to form H<sub>2</sub>, so the Tafel reaction becomes the rate-determining step for the H<sub>2</sub> production ( $\text{H}_{\text{ads}} + \text{H}_{\text{ads}} \rightarrow \text{H}_2$ ). In reverse, if the coverage of H<sub>ads</sub> is low the Heyrovsky reaction becomes the rate-determining step, where a single H<sub>ads</sub> atom attracts both a proton and an electron (e<sup>-</sup>) to produce an H<sub>2</sub> molecule ( $\text{H}_{\text{ads}} + \text{e}^- + \text{H}_3\text{O}^+ \rightarrow \text{H}_2 + \text{H}_2\text{O}$ ). When the Volmer step is the rate-limiting step, the Tafel slope reaches 120 mV dec<sup>-1</sup>. Whereas the Tafel slope is reduced to 40 or 30 mV dec<sup>-1</sup> when the Heyrovsky step or Tafel step is the rate-limiting step.<sup>52,53</sup> Fig. 4c presents the Tafel plots of all samples, where V<sub>2</sub>C/BVO shows a smaller Tafel slope of 160 mV dec<sup>-1</sup> than those of V<sub>2</sub>C/Bi (173 mV dec<sup>-1</sup>) and V<sub>2</sub>C (204 mV dec<sup>-1</sup>), revealing that the HER process of V<sub>2</sub>C/BVO is controlled by the Volmer–Heyrovsky mechanism and the Volmer step is the rate-determining step.<sup>54,55</sup> As previously reported, under acidic conditions, the reaction between –OH functional groups on the surface of MXenes and H<sup>+</sup> ions in H<sub>2</sub>SO<sub>4</sub> aqueous solution promotes the current before the HER process, which explains the main reason for the high Tafel slope in the current study.<sup>56,57</sup> To further probe the origin of the enhanced HER performance of prepared electrocatalysts, the EIS measurement is a crucial method to investigate the interfacial properties and the charge-transfer resistance between the solution and the electrode. Meanwhile, the arc radius in the EIS Nyquist plots could be a credible method to estimate the charge transfer rate.<sup>58</sup> It was commonly believed that a smaller arc in the EIS Nyquist plot implied less charge migration resistance.<sup>58–60</sup> As displayed in Fig. 4d, V<sub>2</sub>C/BVO composite displayed a smaller arc radius, indicating faster charge transfer at the electrode/electrolyte interface, leading to enhanced HER performance. The exchange current density ( $j_0$ ) is another important parameter in the evaluation of HER activity, which is defined by the cathodic current density of the Tafel equation at the zero overpotential. The  $j_0$  of all the catalysts was evaluated from the Tafel plot by extending to X-axis (Fig. S5†). The  $j_0$  of the V<sub>2</sub>C/BVO hybrid is estimated to be 16  $\mu\text{A cm}^{-2}$  which is significantly higher than the V<sub>2</sub>C/Bi (10  $\mu\text{A cm}^{-2}$ ) and V<sub>2</sub>C (9  $\mu\text{A cm}^{-2}$ ). It is reflected that the inter-

facial electronic coupling could boost the electron transfer capability of V<sub>2</sub>C/BVO hybrids for HER catalysis.<sup>61</sup> In addition, to better understand the intrinsic activity of the catalyst, we further evaluated the double-layer capacitance ( $C_{\text{dl}}$ ) by a series of cyclic voltammetry (CV) measurements at different scan rates from 10 to 100 mV s<sup>-1</sup> in the capacitive range of 0 to 0.1 V (vs. RHE). As expected, the V<sub>2</sub>C/BVO electrode exhibits a higher current than that of V<sub>2</sub>C and V<sub>2</sub>C/Bi as shown in Fig. S6.† It indicates more accessible active sites are created during hydrothermal synthesis of V<sub>2</sub>C/BVO which is favourable for the enhanced HER activity. To calculate the double layer capacitance ( $C_{\text{dl}}$ ), we first determined the capacitive density current average ( $\Delta j$ ) using the formula  $\Delta j = |j_c - j_a|/2$ , where  $j_c$  represents the cathodic density current and  $j_a$  represents the anodic density current. At a potential of 0.05 V, we plotted  $\Delta j$  against various scan rates and determined the linear slope value, which corresponded to  $C_{\text{dl}}$ .<sup>62</sup> As shown in Fig. 4e, the  $C_{\text{dl}}$  of V<sub>2</sub>C/BVO (6.66 mF cm<sup>-2</sup>) was much larger than that of V<sub>2</sub>C/Bi (3.23 mF cm<sup>-2</sup>) and V<sub>2</sub>C (1.56 mF cm<sup>-2</sup>) which was attributed to the high SSA of the 2D/2D V<sub>2</sub>C/BVO composite and is beneficial to the exposure of more accessible active sites for the HER and diffusion of electrocatalytic active species.<sup>63,64</sup> We can conclude that the BiVO<sub>4</sub> which is *in situ* formed and uniformly coated on the V<sub>2</sub>C layers provides additional active sites for HER reaction. Therefore, the actual active site responsible for the HER reaction in the V<sub>2</sub>C/BVO composite is the interface between the V<sub>2</sub>C and BiVO<sub>4</sub> layers, where synergistic effects may occur.

All the electrochemical measurements for HER reactions show that V<sub>2</sub>C/BVO composite exhibits impressive electrocatalytic ability in acidic media compared to pure V<sub>2</sub>C and V<sub>2</sub>C/Bi composite. The outstanding electrocatalytic performance of the V<sub>2</sub>C/BVO composite is supposed to be originating from; (i) a well-constructed 2D/2D (V<sub>2</sub>C/BVO) structure which is beneficial to increase the SSA, which allows the exposure of numerous catalytic active sites and full penetration of electrolyte, (ii) the synergistic effect of V<sub>2</sub>C and BiVO<sub>4</sub>, and their intrinsic electrocatalytic response, and (iii) the high metallic conductivity of V<sub>2</sub>C MXene sheets acts as a conductive material which facilitates electron transport. All these features confirm that the newly prepared composite (V<sub>2</sub>C/BVO) in a facile hydrothermal way could be considered as a promising non-noble catalyst for HER application. HER efficiencies of as-prepared V<sub>2</sub>C/BVO composite were also evaluated as electrocatalyst for HER in alkaline (1 M KOH, pH = 14) and neutral solution (1 M Na<sub>2</sub>SO<sub>4</sub>, pH = 7). Fig. S7a† shows that V<sub>2</sub>C/BVO composite exhibits a high overpotential of 584 and 679 mV at a current density of 10 mA cm<sup>-2</sup> in basic and neutral media, respectively. The results suggest that the HER reactions of V<sub>2</sub>C/BVO under alkaline conditions are harder than in acidic solutions. This indicates the ability of the V<sub>2</sub>C/BVO catalyst to adsorb hydrogen rather than accelerated water dissociation.

Long-term stability testing on V<sub>2</sub>C/BVO was also carried out by a chronopotentiometry test. Fig. 4f shows low voltage increasing over 8 h at a constant current density (10 mA cm<sup>-2</sup>)

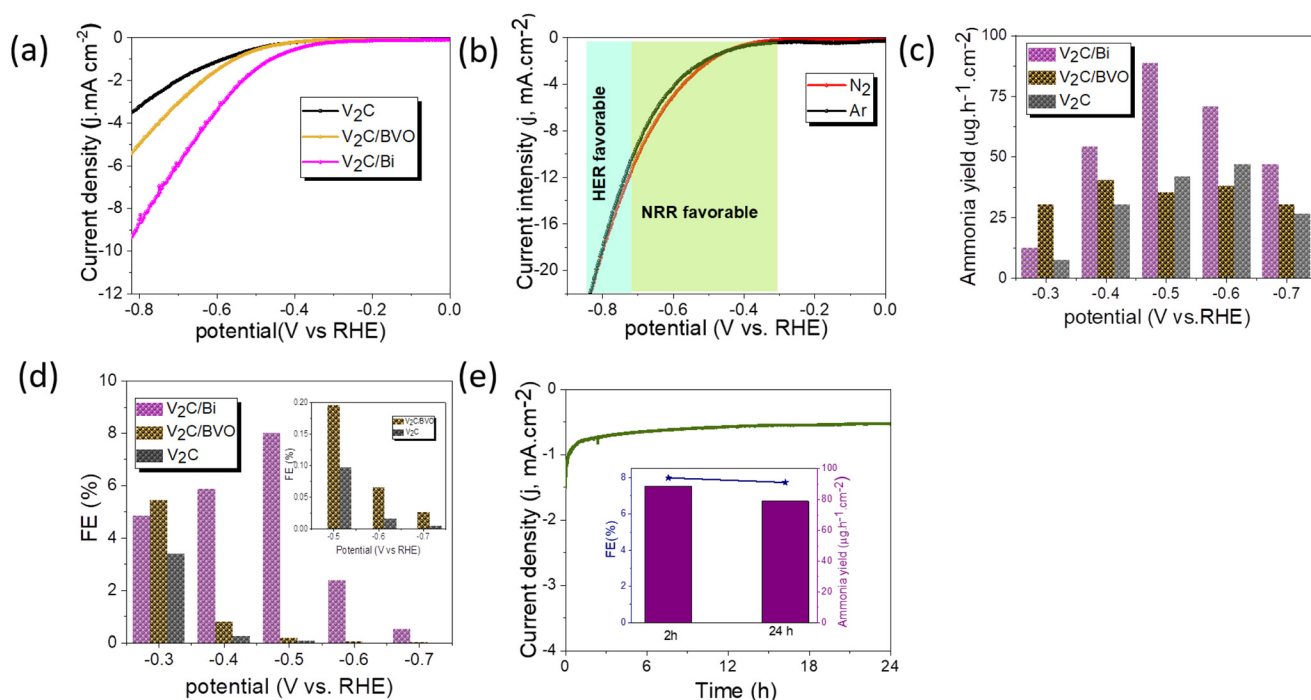


operation. Fig. S7(b and g)† shows the SEM, EDS, and XPS of the catalyst after the stability test. SEM and EDS elemental mapping images show that  $V_2C/BVO$  still has a multilayer structure without being damaged and the existence of all chemical composition as before electrolysis. As well, post-stability XPS measurements indicate the high structural stability of  $V_2C/BVO$  toward the HER process. Furthermore, we conducted the EIS analysis after stability as shown in Fig. S7h.† The Nyquist plots displayed that the arc radius of the  $V_2C/BVO$  composite after and before stability is mostly identical, revealing the same charge transfer capacity and, therefore, no considerable alteration of the electrical properties confirming the high stability of  $V_2C/BVO$ .

### 3.3. Evaluation of NRR performance

The Electrochemical NRR measurements were performed in a two-compartment cell (H-cell) as shown in Fig. S8.† The LSV curves of catalysts were compared under an  $N_2$ -saturated 0.1 M KOH electrolyte (Fig. 5a). It can be found that the current value of  $V_2C/Bi$  is higher than that of  $V_2C$  and  $V_2C/BVO$  at the same potential, suggesting its superior catalytic activity. The result further proves that the addition of the metal Bi effectively improved the catalytic performance of the  $V_2C$ . Linear sweep voltammetry (LSV) measurements of the  $V_2C/Bi$  composite were then performed in Ar- and  $N_2$ -saturated 0.1 M KOH aqueous under room temperature and atmospheric pressure (Fig. 5b). Obviously, the reduction of current density in the potential range of  $-0.4$  V to  $-0.7$  V of  $V_2C/Bi$  in  $N_2$  is slightly lower than that of the Ar atmosphere, implying that NRR was

more favourable over HER in the measured potential window. Next, a CV test was performed between  $-1$  V and  $1.8$  V vs. RHE in  $N_2$ -saturated KOH. Fig. S9a† shows 3 peaks at  $0.45$ ,  $0$ , and  $-0.56$  V vs. RHE. The anodic and cathodic signals at  $0.45$  and  $0$  V vs. RHE can be assigned to the surface reversible oxidation and subsequent reduction of the electrocatalyst. Specifically, these signals indicate the interconversion of species between Bi (0) and Bi (OH)<sub>3</sub> (or BiOOH) during the electrochemical reaction.<sup>65</sup> The broad cathodic peak that appears at  $-0.56$  V is attributed to the reduction of  $Bi^{3+}$  to metallic Bi.<sup>66,67</sup> This observation is consistent with the XRD results, which showed the presence of a small amount of  $Bi_2O_3$  in the  $V_2C/Bi$  composite. The complete reduction of  $Bi^{3+}$  to metallic Bi occurs during the first scan and subsequent scans may not show this cathodic peak due to the saturation of the surface with metallic Bi. After 3 consecutive cycles, the CV curves are almost identical, confirming the stability of the electrode before the NRR test. The electrochemical NRR performances of catalysts were further evaluated by chronoamperometry tests under potentials ranging from  $-0.3$  to  $-0.7$  V vs. RHE for 2 h in  $N_2$ -saturated KOH aqueous solution as shown in Fig. S9b–d.† It is clearly depicted that the current density improved with the increase of potential and remains almost stable during the two hours at different potentials, implying that both  $V_2C/Bi$  and  $V_2C/BVO$  composites exhibit good electrochemical stability during the NRR test. While for  $V_2C$ , at  $-0.5$  V vs. RHE, the current density increased from the initial  $-0.13$  mA cm<sup>-2</sup> value to about  $-1.84$  mA cm<sup>-2</sup> after 2 h, indicating an *in situ* transformation during the electrocatalytic tests. It is possible



**Fig. 5** (a) Comparison of LSV curves of GC electrode,  $V_2C$ ,  $V_2C/BVO$ , and  $V_2C/Bi$  in  $N_2$ -saturated 0.1 M KOH with a scan rate of  $10$  mV s<sup>-1</sup>, (b) LSV curves of  $V_2C/Bi$  in Ar/ $N_2$ -saturated KOH electrolyte with a scan rate of  $10$  mV s<sup>-1</sup>, (c) corresponding  $NH_3$  yields (d) corresponding FE, (e) chronoamperometric test of  $V_2C/Bi$  at  $-0.5$  V for 24 h (insert: corresponding  $NH_3$  yields and FE after stability test).

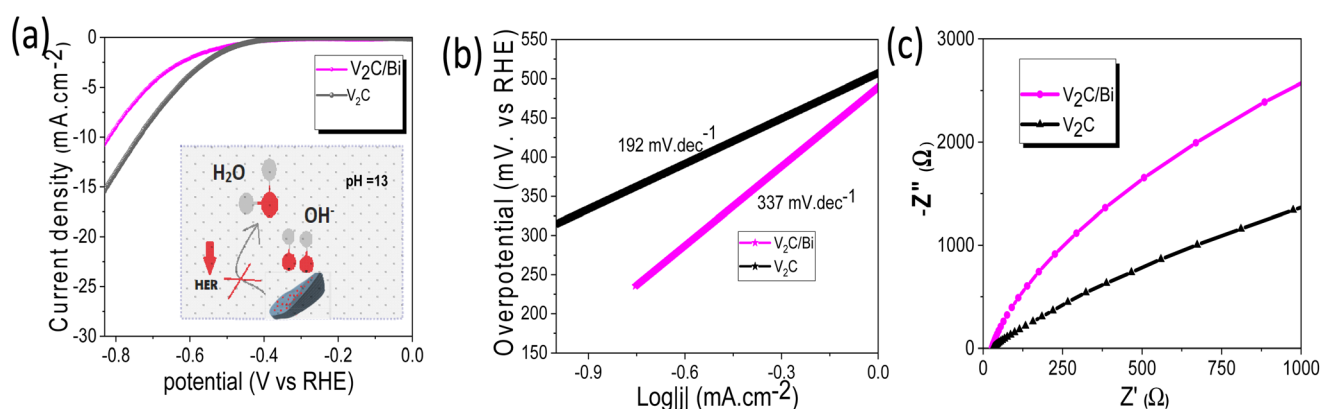


that the decrease in current density observed during the electrocatalytic experiment was due to the oxidation of  $V_2C$  and the formation of  $VO_x$  species. These oxide layers can act as passivating layers that hinder electron transfer between the  $V_2C$  and the electrolyte solution, leading to a decrease in current density over time. After electrolysis at a given potential, the generated ammonia was determined spectrophotometrically by Nessler's method. The  $NH_3$  production and corresponding FE under different potentials are displayed in Fig. 5c and d. As clearly shown, the  $V_2C/Bi$  composite exhibits the highest  $NH_3$  yield of  $88.6 \mu g h^{-1} cm^{-2}$  and the highest faradaic efficiency (FE) of 8% at  $-0.5 V$  vs. RHE which are both higher than the values obtained for  $V_2C/BVO$  ( $NH_3$  yield:  $40.5 \mu g h^{-1} cm^{-2}$ , FE: 5.45%). For confirmation, the indophenol method (Fig. S10d–f†) was also used to measure the concentration of ammonia for  $V_2C/Bi$  at  $-0.5 V$  (V vs. RHE), and the result was  $92 \mu g h^{-1} cm^{-2}$ , which is close to the value calculated from the Nessler's reagent method. This result confirms that both methods produce consistent values.

We compared the ammonia yield rates and the corresponding faradaic efficiencies of  $V_2C/Bi$  with those of the recently reported MXenes-based NRR catalysts. As listed in Table S3,† the ammonia yield produced over  $V_2C/Bi$  is significantly higher than that of all the listed catalysts. Furthermore, we evaluated the NRR performance of the pristine  $V_2C$ . Compared to  $V_2C/Bi$ ,  $V_2C$  exhibits significantly lower NRR activity ( $NH_3$  yield:  $46.8 \mu g h^{-1} cm^{-2}$ , FE: 3.4%), implying that the Bi supported on  $V_2C$  should be the main active sites for NRR for  $V_2C/Bi$ . Obviously, the  $NH_3$  yield, and FE decreased when the applied potentials shifted to more negative values, mainly due to the competitive adsorption between  $N_2$  and  $H_2$  on the electrode surface. Furthermore, no  $N_2H_4$  could be detected after the electrocatalytic NRR at different potentials. This further confirms the outstanding selectivity of the  $V_2C/Bi$  catalyst. To confirm the electrochemical reduction of  $N_2$  to  $NH_3$  on  $V_2C/Bi$ , three controlled experiments were conducted: (i) the working electrode in electrolyte with continual Ar flow at  $-0.5 V$  for 2 h electrolysis; (ii) the working electrode with continual  $N_2$  flow at open circuit potential (OCP) for 2 h elec-

trolysis; and (iii) bare glassy carbon electrode in electrolyte with continual  $N_2$  flow at  $-0.5 V$  for 2 h electrolysis. As shown in Fig. S10,† very low current density and very low  $NH_3$  and corresponding FE were detected for all those control experiments, suggesting that the produced  $NH_3$  is derived from the electrochemical  $N_2$  reduction catalyzed by the  $V_2C/Bi$  catalyst.

For electrocatalytic NRR, long-term catalytic stability is critical for the possible practical application. Fig. 5e shows a very steady current density over 24 h of continuous chronoamperometric test at  $-0.5 V$ , with a 90.5% retention of its initial  $NH_3$  yield and almost unchanged FE after 24 h (inset Fig. 5e). Additionally, the SEM images and corresponding EDS elemental mapping (Fig. S11†) of the  $V_2C/Bi$  after stability revealed that the morphology and the chemical composition have no obvious changes compared to the fresh one, proving the robustness of the heterostructure for stable NRR performance. To further evaluate the long-term stability of  $V_2C/Bi$  catalyst, EIS analysis was recorded before and after 24 h of chronoamperometry analysis. Fig. S11e† indicates that the catalyst exhibits negligible deviation of the arc radius after stability which confirms its excellent long-term stability. Furthermore, the XPS spectra after NRR stability (Fig. S11f–h†) confirm that the  $V_2C/Bi$  well retains its initial chemical bonding state for V 2p and C 1s. The appearance of two new peaks with low intensity at 157.4 and 162.4 eV in the Bi 4f spectra indicates the presence of  $Bi^0$ . As discussed earlier,  $V_2C/Bi$  composite exhibits reversible reduction at 0 V vs. RHE (Fig. S9a†) of bismuth nanoparticles. Thus, these nanoparticles were reduced during the long term experiment to metallic Bi. Subsequently when exposed to air/oxygen, their surface was again oxidized. All the above results unambiguously demonstrate  $V_2C/Bi$  to be a promising NRR catalyst for efficient  $NH_3$  electro-synthesis. Generally, the NRR reaction process involves continuous proton–electron pairs interacting with  $N_2$ . As already proven HER is the main competitive reaction of the NRR. Therefore, increasing  $N_2$  coverage on the electrocatalyst and suppressing the HER process during NRR is an effective strategy to increase the selectivity to  $NH_3$ . Hence, we study the HER activity of the  $V_2C/Bi$  nanocomposite and compare it with that of  $V_2C$



**Fig. 6** (a) LSV of  $V_2C/Bi$  and  $V_2C$  in Ar saturated KOH (insert: the deactivation mechanism of the HER process over  $V_2C/Bi$ ), (b) the corresponding Tafel plots analysis, and (c) EIS of  $V_2C$  and  $V_2C/Bi$ .



MXene. As shown in Fig. 6a,  $V_2C/Bi$  exhibited lower HER performance compared to  $V_2C$ . The Tafel slopes of the samples were then calculated. As demonstrated in Fig. 6b, the Tafel slopes of the  $V_2C/Bi$  composite were higher than that of the pristine  $V_2C$  MXene, indicating a less favourable kinetics process for HER over the  $V_2C/Bi$  electrode. In addition, the EIS measurement (Fig. 6c) shows that the arc radius of  $V_2C/Bi$  is significantly larger than that of  $V_2C$ , indicating low electron transfer in the composite and resulting in lower HER performance. This result confirms the negative effect of the Bi nanoparticles on HER, which, on the other hand, is highly beneficial for the improvement of NRR performance for gradually elevated ammonia yields. Thus, we proved that the HER process can be significantly repressed with Bi by inhibiting the adsorption of H on active sites. The same result has already been proven by previous researchers when studying the NRR activity of the Mo-doped  $V_2C$  system.<sup>68</sup> Theoretical calculations confirmed that  $\Delta G_{N_2^*} > \Delta G_{H^*}$  on Mo-doped  $V_2C$ , which confirms the restriction of the HER reaction after doping of  $V_2C$  with Mo.

In this study, the  $V_2C/Bi$  heterojunction is produced by supporting the Bi-metal on O-terminated  $V_2C$ , indicating that  $V_2C$  can offer electrons that can be transported to Bi through the heterojunction interface (Fig. S12†). The electrons provided by  $V_2C$  are mainly accumulated at the  $V_2C/Bi$  interface. Such electron-rich interfaces facilitate the downshift of the p-band centre of interfacial Bi atoms,<sup>69</sup> which serve as the main active sites to promote the  $N_2$  activation and hydrogenation while impeding the competing hydrogen evolution. This suggests the outstanding selectivity of  $V_2C/Bi$  for  $N_2$ -to- $NH_3$  conversion.

## 4. Experimental procedures

### 4.1 Synthesis of layered $V_2C$

Multilayered  $V_2C$  MXene powder was obtained by etching Al from  $V_2AlC$  powder according to a previous report.<sup>70</sup> Briefly, 6 g NaF was dissolved in 120 mL of deionized water, then 140 mL HCl solution (6 mol  $L^{-1}$ ) was added to obtain the etching solution. After that, 6 g of as-received  $V_2AlC$  MAX phase powder was slowly added to the etching solutions in Teflon-lined stainless-steel autoclaves (500 mL). The Teflon bottle was kept in an oil bath with magnetic stirring at 90 °C for 5 days. After the end of the etching process, the dispersion was then centrifuged at 5000 rpm for 5 min to harvest the sediment, which was further centrifuged and washed with deionized (DI) water and ethanol six times to raise the pH of the supernatant to almost neutral (pH  $\sim$  7). Finally, the solid product was collected and then placed in a vacuum freeze dryer at  $-50$  °C for 24 h to remove the residual water and obtain a powder of  $V_2C$  MXene.

### 4.2 Synthesis of $V_2C/BVO$ by hydrothermal method

The composite  $V_2C/BVO$  was synthesized through a facile hydrothermal method (Fig. 7). To do this, 100 mg of the prepared  $V_2C$  MXene powder was put in 40 mL of deionized water

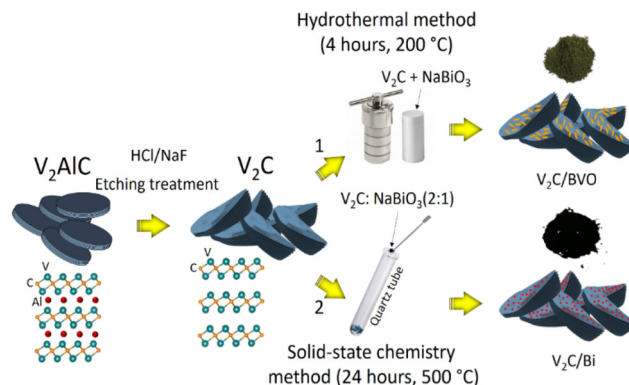


Fig. 7 A schematic illustration of the synthesis of composites  $V_2C/BVO$  and  $V_2C/Bi$ .

and ultrasonically dispersed for 30 minutes to obtain a uniform MXene suspension. Another solution was prepared by dissolving 100 mg  $NaBiO_3$  in 25 mL of deionized water. Then, the  $NaBiO_3$  aqueous solution was slowly added to the uniform MXene suspension. The mixed suspension was vigorously stirred for 30 minutes, then transferred to a 100 mL Teflon-lined autoclave for 4 h at 200 °C. The dark green precipitate was centrifuged and thoroughly washed with deionized water and absolute ethanol several times. The  $V_2C/BVO$  composite was finally obtained by drying it in a vacuum oven at 60 °C overnight.

### 4.3 Synthesis of $V_2C/Bi$ by solid-state annealing technique

The composite  $V_2C/Bi$  was synthesized through a solid-state annealing method following the steps described in the previous report (Fig. 7).<sup>71</sup> In this process, the obtained  $V_2C$  MXene and  $NaBiO_3$  were mixed and crushed for 20 minutes with an agate mortar. After uniformly mixing the two components, the mixture powder was inserted inside a quartz ampoule and then was vacuum sealed to a pressure of approximately  $10^{-5}$  mbar using a vacuum pump. The vacuum sealing of the ampoule was required to prevent the mixture from moisture. Finally, the sealed ampoule including the mixture was placed in a furnace and heated at 500 °C for 24 h using a 5 °C  $min^{-1}$  heating and cooling rate. After annealing, the ampoule was removed from the furnace and cooled in the air. Then, it was opened, and the black precipitate was manually collected.

### 4.4 Characterization

X-ray diffraction (XRD) patterns of the synthesized materials were acquired on a Bruker D8 diffractometer operating at 40 kV with Cu  $K\alpha$  radiation ( $\lambda = 1.54056$  Å). The diffraction patterns were collected between 5° and 60° with a step size of 0.02°. Scanning electron microscopy (SEM) was performed by a field-emission scanning electron microscope (FESEM, Tescan Lyra dual microscope) with attached energy-dispersive X-ray spectroscopy of a 20 mm<sup>2</sup> SDD detector (Oxford Instruments). Transmission electron microscopy (TEM, EFTEM Jeol 2200 FS microscope) equipped with energy dispersive spectrometry



(EDS) at an acceleration voltage of 200 kV. Raman spectra were performed by Raman spectrometer (Renishaw, England) in backscattering geometry with a CCD detector and a DPSS laser (532 nm, 50 mW) with an applied power of 0.5 mW. The samples were prepared by drop-casting a film on a nickel substrate, directly prior to the testing. X-ray photoelectron spectroscopy (XPS) analysis was recorded using a SPECS spectrometer equipped with an Al K $\alpha$  X-ray source (1486.7 eV). The Brunauer-Emmett-Teller (BET) and pore size distribution of the samples were measured by a Micro-metrics ASAP 2020 apparatus at liquid nitrogen temperature.

#### 4.5. Electrochemical measurements

**4.5.1. Electrochemical HER measurements.** All electrochemical measurements were carried out in a three-electrode system and a PGSTAT 204 (Utrecht, Netherlands, NOVA Version 2.1.4) electrochemical workstation at room temperature using Ag/AgCl as the reference electrode, carbon rod as the counter electrode, and the glassy carbon electrode as a working electrode. Initially, prior to the electrochemical measurements, the glassy carbon electrode (GC) was polished with an alumina suspension and then subjected to ultrasonic cleaning in double distilled water for 2 minutes. The working electrode was prepared as follows: 2 mg sample, 20  $\mu$ L of Nafion solution (5 wt%), 100  $\mu$ L of ethanol, and 300  $\mu$ L DI water were dispersed by sonication for 30 minutes. A dispersed ink of 10  $\mu$ L was coated onto the surface of a working electrode using the drop-casting method and dried under a vacuum. The Ag/AgCl reference electrode was calibrated with respect to a reversible hydrogen electrode (RHE) according to the Nernst equation:

$$E_{\text{RHE}} = E_{\text{Ag/AgCl}} + 0.197 + 0.059 \text{ pH}$$

Linear sweep voltammetry (LSV) was performed in 0.5 M H<sub>2</sub>SO<sub>4</sub> with a sweep rate of 5 mV s<sup>-1</sup>. The Tafel slope of the catalyst can be calculated by the formula  $\eta = a + b \log|j|$  ( $\eta$ ,  $a$ ,  $b$ , and  $j$  are standing for overpotential, Tafel constant, Tafel slope, and current density, respectively). The overpotential for HER was calculated using the following equation  $\eta = 0 - E$  (vs. RHE (V)). Electrochemical impedance spectroscopy (EIS) was recorded by applying an AC potential amplitude of 10 mV within the frequency range from 0.1–1000 Hz. The double-layer capacitances were obtained through cyclic voltammetry (CV) with various scan rates in the voltage range of 0–0.1 V vs. RHE.

**4.5.2. Electrochemical NRR measurements.** Electrochemical measurements were performed with an Autolab PGSTAT 204 (Metrohm, Switzerland) in a two-compartment cell (H-cell) electrocatalytic system separated by a proton-exchange membrane (Nafion 211) at room temperature and atmospheric pressure. A platinum plate, Ag/AgCl/saturated KCl, and glassy carbon (effective area: 0.158 cm<sup>2</sup>) were used as a counter electrode, reference electrode, and working electrode, respectively. The working electrode was polished and prepared following the above-mentioned steps. All potentials in this work were converted to the reversible hydrogen electrode (RHE) using the Nernst equation ( $E_{\text{RHE}} = E_{\text{Ag/AgCl}} + 0.197 + 0.059 \text{ pH}$ ). Before

performing the NRR test, a Nafion membrane was pretreated in H<sub>2</sub>O<sub>2</sub> (5 wt%) aqueous solution at 80 °C and then in ultra-pure water at 80 °C each for 1 hour. For NRR experiments, 0.1 M KOH (50 mL in each cell compartment) was purged with N<sub>2</sub> (99.99%) using a constant flow of 20 mL min<sup>-1</sup> during the whole electrolytic process, and a 40 mL acid trap (0.1 M H<sub>2</sub>SO<sub>4</sub>) was connected to both compartments. The chronoamperometry tests were conducted at a series of applied potentials (−0.3, −0.4, −0.5, −0.6, and −0.7 V vs. RHE) for 2 hours in ambient conditions. The LSV curves were collected in 0.1 M KOH aqueous solution saturated with Ar or N<sub>2</sub>, at a sweep rate of 100 mV s<sup>-1</sup>. A specific amount of ammonia (NH<sub>3</sub>) produced in the acid trap was measured by HI83300 multi-parameter photometer (Hanna Instruments, Woonsocket, Rhode Island, USA). The ASTM D1426 Nessler method which has been noted in the previous report,<sup>26</sup> was adopted to determine the concentration of produced NH<sub>3</sub>. Briefly, a cuvette filled with 10 mL of unreacted sample was placed in the photometer to obtain the background signal. Subsequently, four drops of ammonia low-range reagent A and ammonia low-range reagent B were added to the sample, then mixed thoroughly, and measured the absorbance of the reacted sample was at 420 nm.

The ammonia formation rate was calculated according to eqn (1):

$$\text{NH}_3 = \frac{C_{\text{NH}_3} \times V}{t \times A} \quad (1)$$

where  $C_{\text{NH}_3}$  is the measured NH<sub>3</sub> concentration,  $V$  is the volume of either the cathode or the acid trapped,  $t$  is the reduction reaction time and  $A$  is the area of the working electrode (0.158 cm<sup>2</sup>).

The faradaic efficiency was calculated based on eqn (2):

$$\text{FE} = \frac{3F \times C_{\text{NH}_3} \times V}{\text{MW}_{\text{NH}_3} \times Q} \% \quad (2)$$

where  $F$  is the Faraday constant,  $\text{MW}_{\text{NH}_3}$  is the molecular weight of ammonia (17 g mol<sup>-1</sup>),  $Q$  is the total charge of the electrode, and  $C_{\text{NH}_3}$  and  $V$  are as above explanations.

The amount of Hydrazine (N<sub>2</sub>H<sub>4</sub>) after the electrolysis reaction was detected by the D1385 method.<sup>26</sup> For determination of the generated hydrazine, 10 mL of the electrolyte after the NRR test was taken out and twelve drops of hydrazine reagent were subsequently added. The absorbance was measured at a wavelength of 466 nm, after 12 minutes.

## 5. Conclusions

Two nanocomposites consisting of 2D/2D V<sub>2</sub>C/BVO and 2D/0D V<sub>2</sub>C/Bi NPs were successfully prepared and applied for electrocatalytic HER and NRR. The V<sub>2</sub>C/BVO composite has been synthesized *via* a facile hydrothermal method and applied as the catalyst electrode for HER in acidic conditions. Compared to pristine V<sub>2</sub>C, the HER activity of V<sub>2</sub>C/BVO has been greatly increased due to the strong interfacial interactions between BVO sheets and V<sub>2</sub>C MXene nanosheets. This 2D/2D structure



provided sufficient active sites and ensured favorable electrochemical kinetics, improving the HER activity. Furthermore, the optimized composite showed excellent long-term stability over 8 h in acidic media at the current density of 10 mA cm<sup>-2</sup>. Also, V<sub>2</sub>C/Bi was proven to be a highly efficient NRR electrocatalyst, achieving a FE and NH<sub>3</sub> yield of 8% and 88.6 µg cm<sup>-2</sup> h<sup>-1</sup>, respectively. The obtained values are among the highest NRR selectivity for MXenes-based NRR catalysts. We also proved that impeding the competing HER by doping V<sub>2</sub>C with p block metal (Bi) is an interesting strategy to achieve high N<sub>2</sub>-to-NH<sub>3</sub> conversion.

This work provides inexpensive strategies for fabricating novel V<sub>2</sub>C-based catalysts and may shed some light on energy-related applications.

## Conflicts of interest

The authors declare no competing financial interest.

## Acknowledgements

S. A. would like to acknowledge the European Structural and Investment Funds, "CHEMFELLS IV" (CZ.02.2.69/0.0/0.0/20\_079/0017899). This work was supported by ERC-CZ program (project LL2101) from Ministry of Education Youth and Sports (MEYS).

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