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# Molten salt-mediated electrocatalytic hydrogen evolution†

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**We report the synthesis of MoS<sub>2</sub> nanosheet-supported Rh nanoclusters via a facile yet efficient molten salt electrolysis method followed by wet chemical reduction for electrocatalytic hydrogen evolution. In-depth study reveals that the uniform Rh nanoclusters on MoS<sub>2</sub> nanosheets can shorten the electronic transport path, and effectively promote the hydrogen evolution reaction kinetics.**

With the continuous development of wind and solar power generation, the use of green electricity to produce hydrogen has become a research hotspot. Electrolysis of water splitting is an effective method for high-purity hydrogen production.<sup>1–3</sup> However, this process consumes excessive energy due to the slow hydrogen evolution reaction (HER) kinetics.<sup>4–7</sup> Therefore, there is an urgent need to discover cathode catalysts with low overpotential and high activity in generating hydrogen. Platinum is currently recognized as the optimal choice for HER catalysis under acidic conditions. However, the limited reserves and high price hinder its widespread applications. Additionally, acidic conditions are more favorable for the HER compared to neutral and alkaline ones.<sup>8–10</sup> Unfortunately, most transition metals are not stable under acidic conditions. Consequently, much effort has been devoted to designing affordable and stable cathode catalysts that can perform well under alkaline or neutral conditions.

Molybdenum disulfide (MoS<sub>2</sub>) is promising as a HER catalyst due to their advantages such as tunability, low cost, and chemical stability.<sup>11–14</sup> However, the catalytic performance of pristine molybdenum disulfide is not satisfactory mainly because of in-plane inertness and limited edge active sites.<sup>15</sup> Therefore, various strategies have been developed to enhance its catalytic activity. In particular, the decoration of MoS<sub>2</sub> nanosheets with HER-active components holds great promise since this approach could not only restrict the restacking between nanosheets, but greatly increase the number of exposed active sites. Rhodium (Rh) could serve as an ideal candidate of the active components to modify MoS<sub>2</sub> nanosheets because of its HER activity like that of Pt and much higher corrosion resistance under harsh alkaline/acidic conditions than that of Pt. Currently, there are still great challenges to fabricate the hybrids of Rh-modified MoS<sub>2</sub> nanosheets for high-performance and low-cost HER catalysis: the synthesis techniques for MoS<sub>2</sub> nanosheets, including the hydrothermal method and chemical vapor deposition, frequently exhibit low production yields, require long time, and are carried out in closed reaction systems, and thus it is crucial to explore high-throughput methods for producing MoS<sub>2</sub> nanosheets; in addition, Rh is expensive and scarce in nature, and it is therefore essential to significantly improve its utilization for greatly enhancing the activity of MoS<sub>2</sub> nanosheets, while reducing its loading.

The molten salt electrolysis could be a highly promising environment-friendly strategy to rapidly synthesize MoS<sub>2</sub> nanosheets with high product yields.<sup>16</sup> The high-temperature environment can further greatly accelerate the chemical reaction rate and thereby enhance the synthesis efficiency. Rh in the form of nanoclusters could significantly increase its dispersion and facilitate its interfacial charge transfer through the Rh/MoS<sub>2</sub> interfaces;<sup>17</sup> therefore, employing Rh nanoclusters to decorate the MoS<sub>2</sub> nanosheets may effectively improve the catalytic activity with extremely low precious metal content. In this study, for the first time, we synthesized MoS<sub>2</sub> nanosheet-supported Rh nanoclusters (MoS<sub>2</sub> NS@Rh NC) via the facile yet efficient molten salt electrolysis method for application as an

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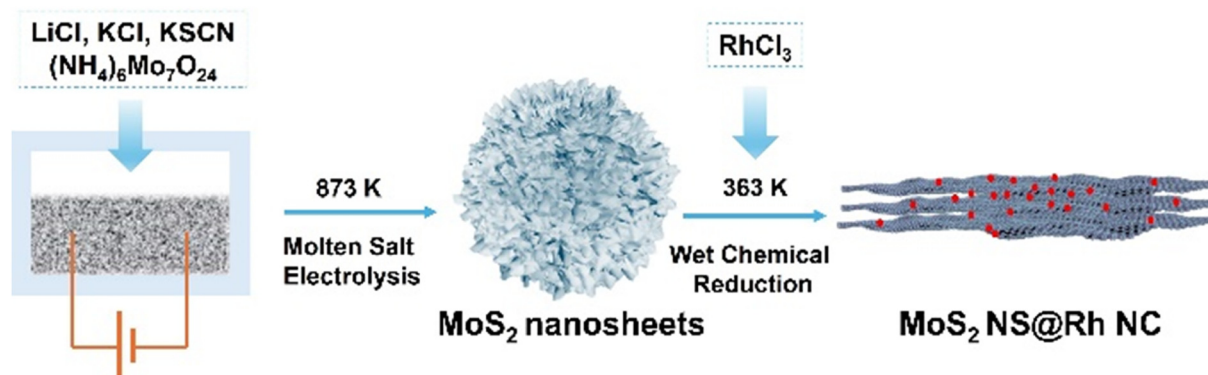


Fig. 1 Schematic illustration of the synthesis of  $\text{MoS}_2\text{ NS@Rh NC}$ .

electrocatalyst toward the HER. It is discovered that the uniform Rh nanoclusters on the  $\text{MoS}_2$  nanosheets obtained using 3.5% Rh can effectively promote the HER. This catalyst exhibits a small overpotential of 27 mV at  $10\text{ mA cm}^{-2}$ , which is close to that of commercial Pt/C (12 mV). In addition, it shows a small Tafel slope of  $47\text{ mV dec}^{-1}$  with good stability.

The synthesis process of  $\text{MoS}_2\text{ NS@Rh NC}$  is schematically shown in Fig. 1.  $\text{LiCl}$ ,  $\text{KCl}$ , anhydrous  $(\text{NH}_4)_6\text{Mo}_7\text{O}_{24}$ , and anhydrous  $\text{KSCN}$  were directly introduced into the reactor. The electrolysis was then carried out at a temperature of  $873\text{ K}$  with a current density of  $1.55\text{ A cm}^{-2}$  for only 1 h. The black precipitate obtained was rinsed thoroughly with distilled water and anhydrous ethanol multiple times. An appropriate volume of  $\text{RhCl}_3$  solution ( $1\text{ mg mL}^{-1}$ ) was subsequently added into the suspension and this reaction system was maintained at  $363\text{ K}$  for 30 min.

The XRD pattern of  $\text{MoS}_2\text{ NS@Rh NC}$  (Fig. 2A) shows characteristic peaks at  $14.4^\circ$ ,  $32.7^\circ$ ,  $33.5^\circ$ ,  $35.9^\circ$ ,  $39.6^\circ$ ,  $49.8^\circ$ ,  $58.4^\circ$ , and  $60.4^\circ$ , which correspond to (002), (100), (101), (102), (103), (105), (110), and (112) planes of hexagonal  $\text{MoS}_2$  (ICCD#01-077-1716; space group,  $P6_3$ ;  $a_0 = b_0 = 3.1602\text{ \AA}$ ;  $c_0 = 12.2940\text{ \AA}$ ). There is no peak from Rh, which could be due to the

too low content of Rh in the sample  $\text{MoS}_2\text{ NS@Rh NC}$ . The peaks for the high-resolution XPS spectrum of  $\text{MoS}_2\text{ NS@Rh NC}$  were deconvoluted by referring to previous literature.<sup>18</sup> The Rh 3d can be deconvoluted into peaks at 308.2, 309.2, and 313.2 eV, which are ascribed to  $\text{Rh}^{1+}$ ,  $\text{Rh}^{2+} 3d_{5/2}$ , and  $\text{Rh}^{2+} 3d_{3/2}$ , respectively.<sup>19</sup> The high-resolution XPS spectrum of Mo 3d shows  $\text{Mo}^{4+}$  signals located at 229.8 and 233.2 eV of  $\text{Mo} 3d_{5/2}$  and  $\text{Mo} 3d_{3/2}$ , accompanied by the relatively weak  $\text{Mo}^{6+}$  signals of 236.9 and 233.9 eV arising from the surface oxidation of  $\text{MoS}_2$  nanosheets.<sup>17</sup> The peak at 226.9 eV belongs to S 2s. Fig. 2D shows three peaks at 162.6 eV 163.8 eV and 169.7 eV, which can be assigned to S  $2p_{3/2}$  and S  $2p_{1/2}$  of  $\text{S}^{2-}$  and the S species in  $\text{MoS}_2$ .<sup>20,21</sup>

The scanning electron microscopy (SEM) image of  $\text{MoS}_2\text{ NS@Rh NC}$  (Fig. 3A) shows a flower-like nanostructure assembled from interconnected nano-flakes. The SEM-EDX elemental mapping images of  $\text{MoS}_2\text{ NS@Rh NC}$  (Fig. 3B) clearly indicate that the Mo (red color), S (green color), and Rh (blue color) elements are distributed homogeneously in Rh- $\text{MoS}_2$ . The transmission electron microscopy (TEM) image further indicates that the  $\text{MoS}_2\text{ NS@Rh NC}$  exhibits a typical nanosheet structure, which is beneficial for its contact with the electrolyte. The high-magnification TEM image (Fig. 3D) reveals that Rh nanoclusters are evenly distributed on the surface of the  $\text{MoS}_2$  nanosheets. The TEM images for the samples of 7.0%  $\text{MoS}_2\text{ NS@Rh NC}$  and 2.8%  $\text{MoS}_2\text{ NS@Rh NC}$  are displayed in Fig. S2 (ESI<sup>†</sup>). It shows that the Rh nanocluster's size and the intensity of the dispersion of 7.0%  $\text{MoS}_2\text{ NS@Rh NC}$  are larger than that of 2.8%  $\text{MoS}_2\text{ NS@Rh NC}$ . The high-resolution transmission electron microscopy (HRTEM) image (Fig. 3E) shows clear lattice fringes with a spacing of 0.61 nm, which corresponds to the (002) plane of hexagonal  $\text{MoS}_2$ . The crystalline fringes can be observed with the lattice spacing of 0.23 nm (Fig. S4, ESI<sup>†</sup>), which can be assigned to the (103) plane. Fig. S5 (ESI<sup>†</sup>) shows the adsorption-desorption and pore size distribution curves. The sample has a BET surface area of  $173.23\text{ m}^2\text{ g}^{-1}$  with the average pore diameter of 9.69 nm, which offers rich space for the adsorption of water molecules and desorption of hydrogen molecules.

The standard three-electrode system was employed to evaluate the HER performance of the  $\text{MoS}_2\text{ NS@Rh NC}$  electrode.

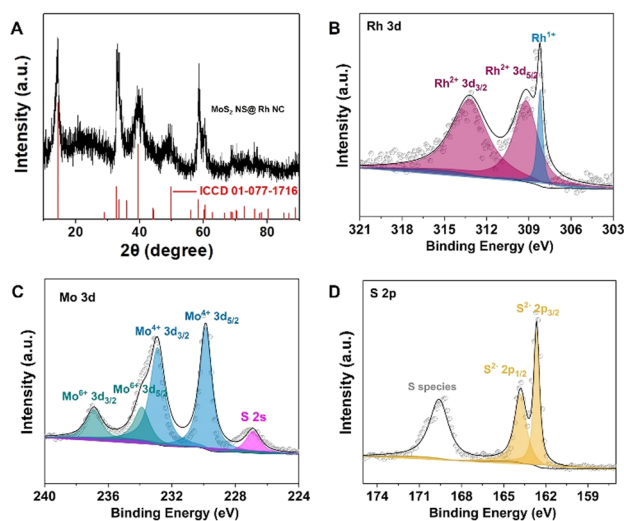


Fig. 2 The XRD image of  $\text{MoS}_2\text{ NS@Rh NC}$  (A) and XPS spectra of Rh 3d (B), Mo 3d (C), and S 2p (D).

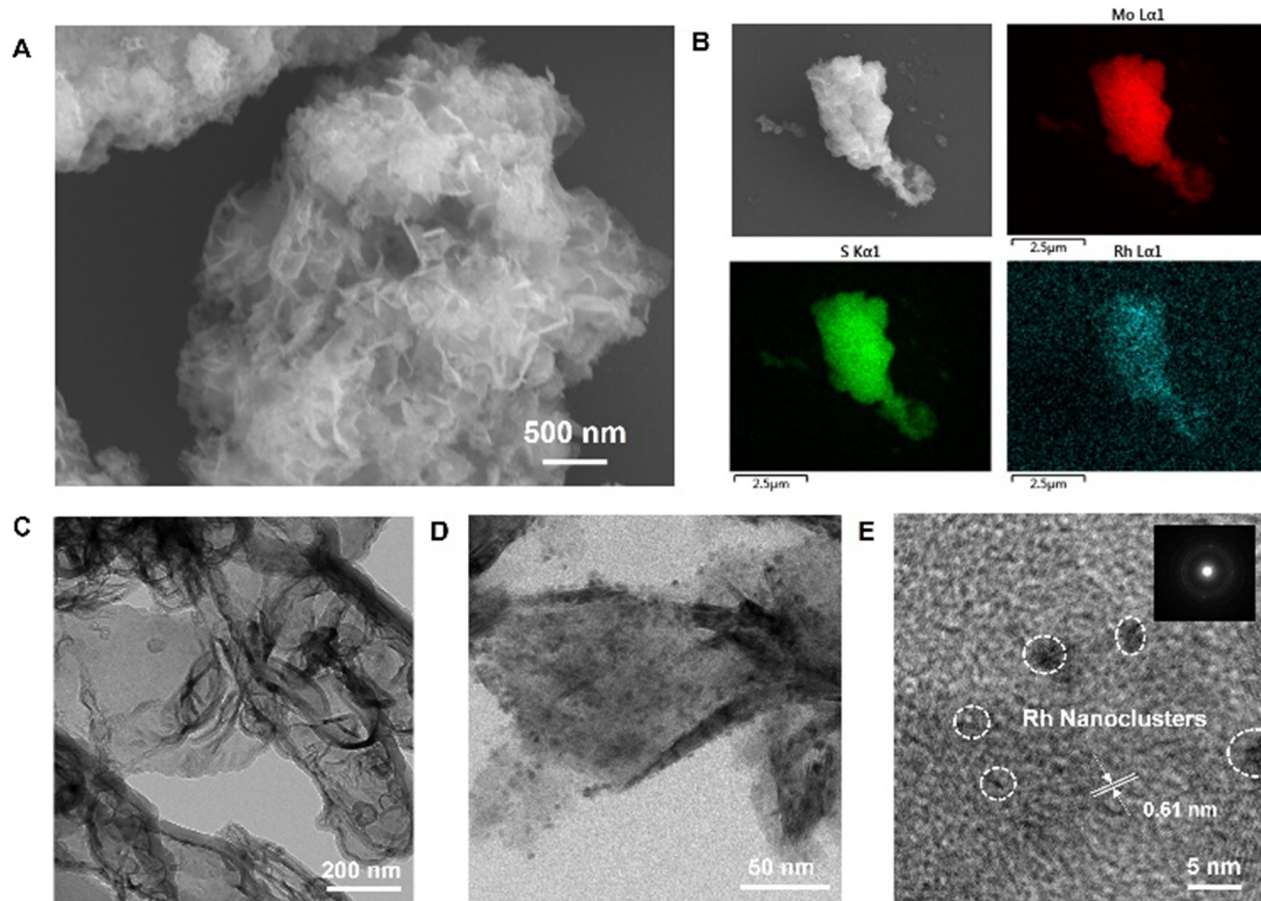


Fig. 3 SEM image of MoS<sub>2</sub> NS@Rh NC (A), energy-dispersive X-ray (EDX) results (B), and TEM (C and D) and HRTEM (E) images of MoS<sub>2</sub> NS@Rh NC.

The LSV curves (Fig. 4A) show that the 3.5 wt% MoS<sub>2</sub> NS@Rh NC electrode has a small overpotential of 27 mV at 10 mA cm<sup>-2</sup>, which is smaller than that of MoS<sub>2</sub> (308 mV), and close to the catalyst of 20% Pt/C (12 mV). The Tafel slope is an important evaluation index for the catalytic reaction kinetics. The Tafel plots (Fig. 4B) show that the Tafel slope of 3.5 wt% MoS<sub>2</sub> NS@Rh NC (47 mV dec<sup>-1</sup>) is close to that of 20% Pt/C (36 mV dec<sup>-1</sup>), and much smaller than that of MoS<sub>2</sub> (142 mV dec<sup>-1</sup>).

The Tafel slope values of the above samples indicate that the introduction of Rh nanoclusters can effectively enhance the HER kinetics of MoS<sub>2</sub>, because the uniform dispersion of Rh can provide a more convenient channel for electron transfer. In addition, the content of Rh has been tailored for investigating the influence of Rh content on HER performance. As shown in Fig. 4C, with the increase in the Rh content, the HER performance of the catalyst is improved significantly. When the content of Rh reaches 3.5%, the HER performance is close to that when the content is 7.0%. Therefore, the content of 3.5% Rh is the ideal content. The Tafel slope (Fig. 4D) further indicates that the reaction kinetics of the 3.5% MoS<sub>2</sub> NS@Rh NC electrode is very similar to that of 7.0% MoS<sub>2</sub> NS@Rh NC. Electrochemical impedance spectroscopy (EIS) was further employed to explore the HER kinetics. The EIS data (Fig. 4E) show that the 3.5% MoS<sub>2</sub> NS@Rh NC electrode has the smallest

electron transfer resistance, indicating the fastest HER on the surface of 3.5% MoS<sub>2</sub> NS@Rh NC. Stability is another important factor in evaluating catalysts. The chronopotentiometric curve of the 3.5% MoS<sub>2</sub> NS@Rh NC electrode at 10 mA cm<sup>-2</sup> (Fig. 4F) shows negligible change in the potential, indicating its good stability. For further evaluation of morphology stability, TEM (Fig. S6, ESI<sup>†</sup>) analysis was performed on the sample after stability testing. It was found that the test pieces still maintained their original nanosheets with a dispersed nanocluster structure after the stability testing, exhibiting good morphology stability. The XRD (Fig. S7, ESI<sup>†</sup>) stability testing shows that the 3.5% MoS<sub>2</sub> NS@Rh NC retains the stable phase structure. In addition, the XPS (Fig. S8, ESI<sup>†</sup>) after stability testing indicates that the Rh<sup>2+</sup>, Mo<sup>6+</sup>, and S<sup>0</sup> species almost disappear, which indicates that the valences of the catalysts have changed during the electroreduction process.

The superior HER performance of MoS<sub>2</sub> NS@Rh NC could be ascribed to the following reasons: first, the 3D interconnected nanosheet structure offers numerous exposed active sites for adsorption of water molecules and desorption of hydrogen molecules; second, the Rh nanoclusters uniformly assembled on the surface of MoS<sub>2</sub> nanosheets shorten the electron transport paths and enhances the intrinsic activity, leading to enhanced reaction kinetics; third, the highly active



Fig. 4 LSV curves (A) and Tafel plots (B) of different samples. LSV curves (C) and Tafel plots (D) of MoS<sub>2</sub> NS@Rh NC with different Rh contents. The Nyquist plots (E) of MoS<sub>2</sub> NS@Rh NC with different Rh contents. The chronopotentiometry (CP) image (F) of MoS<sub>2</sub> NS@Rh NC for 10 hours.

Rh nanoclusters anchored on the planes of the MoS<sub>2</sub> nanosheets significantly enhance the in-plane HER activity.

In summary, MoS<sub>2</sub> nanosheet-supported Rh nanoclusters were synthesized *via* a facile, yet efficient molten salt electrolysis method followed by wet chemical reduction, which can be easily implemented for large-scale synthesis. The uniform Rh nanoclusters on MoS<sub>2</sub> nanosheets can enhance the intrinsic activity of the catalysts, as well shorten the electron transport path, and effectively promote the HER kinetics. This study not only creates a highly efficient and inexpensive HER catalyst for water splitting but provides a novel strategy to synthesize unique nanostructured materials for important applications such as catalysis, energy storage, the environment, and biomedicine.

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## Data availability

The data supporting this article have been included as part of the ESI.†

## Conflicts of interest

The authors declare no competing financial interest.

## References

- W. Hu, Q. Yan, X. Wang, J. Lu, Q. He, Q. Zhang and W. Yuan, *Adv. Funct. Mater.*, 2024, 2411904.
- X. Liu, B. Li, F. A. Soto, X. Li, R. R. Unocic, P. B. Balbuena, A. R. Harutyunyan, J. Hone and D. V. Esposito, *ACS Catal.*, 2021, 11, 12159–12169.
- W. Han, J. Ning, Y. Long, J. Qiu, W. Jiang, Y. Wang, L. A. Shah, D. Yang, A. Dong and T. Li, *Adv. Energy Mater.*, 2023, 13, 2300145.
- H. Xiong, X. Zhang, X. Peng, D. Liu, Y. Han and F. Xu, *Chem. Commun.*, 2024, 60, 11112–11115.
- Y. Li, J. Han, W. Bao, J. Zhang, T. Ai, M. Yang, C. Yang and P. Zhang, *J. Energy Chem.*, 2024, 90, 590–599.
- L. A. Zavala, K. Kumar, V. Martin, F. Maillard, F. Maugé, X. Portier, L. Olivier and L. Dubau, *ACS Catal.*, 2023, 13, 1221–1229.
- S. Qiang, Z. Li, S. He, H. Zhou, Y. Zhang, X. Cao, A. Yuan, J. Zou, J. Wu and Y. Qiao, *Nano Energy*, 2025, 134, 110564.
- X. Guo, E. Song, W. Zhao, S. Xu, W. Zhao, Y. Lei, Y. Fang, J. Liu and F. Huang, *Nat. Commun.*, 2022, 13, 5954.
- Y. Wang, W. Luo, S. Gong, L. Luo, Y. Li, Y. Zhao and Z. Li, *Adv. Mater.*, 2023, 35, 2302499.
- Y. Zhang, Z. Li, S. He, Y. Qiao, A. Yuan, J. Wu and H. Zhou, *J. Colloid Interface Sci.*, 2025, 679, 20–29.
- R. Li, J. Liang, T. Li, L. Yue, Q. Liu, Y. Luo, M. S. Hamdy, Y. Sun and X. Sun, *Chem. Commun.*, 2022, 58, 2259–2278.
- J. Xu, A. Gulzar, Y. Liu, H. Bi, S. Gai, B. Liu, D. Yang, F. He and P. Yang, *Small*, 2017, 13, 1701841.
- S. Ma, Q. Yan, H. Gui, R. Gao, L. Zhang, X. Bai and K. Cheng, *New J. Chem.*, 2025, 49, 84–91.
- J. Lim, S. J. Heo, M. Jung, T. Kim, J. Byeon, H. Park, J. E. Jang, J. Hong, J. Moon, S. Pak and S. Cha, *Small*, 2024, 20, 2402272.
- Y. Zhang, X. Wang, X. Song and H. Jiang, *Nano Res.*, 2024, 17, 9507–9517.
- J. Luo, Q. Qing, Z. Wang, S. Liu, J. Chen and Y. Lu, *Sep. Purif. Technol.*, 2024, 332, 125829.
- Y. Cheng, S. Lu, F. Liao, L. Liu, Y. Li and M. Shao, *Adv. Funct. Mater.*, 2017, 27, 1700359.
- J. Chen, Y. Zhuang, Y. Qiao, Y. Zhang, A. Yuan and H. Zhou, *Int. J. Miner., Metall. Mater.*, 2025, 32, 476–487.
- L. Yang, G. Li, R. Ma, S. Hou, J. Chang, M. Ruan, W. Cai, Z. Jin, W. Xu, G. Wang, J. Ge, C. Liu and W. Xing, *Nano Res.*, 2021, 14, 2853–2860.
- X. Meng, C. Ma, L. Jiang, R. Si, X. Meng, Y. Tu, L. Yu, X. Bao and D. Deng, *Angew. Chem., Int. Ed.*, 2020, 59, 10502–10507.
- F.-Y. Yu, H. Sun, H.-Q. Tan, Y.-G. Li, S.-T. Lee and Z.-H. Kang, *Chem. Catal.*, 2024, 4, 101002.