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Blue quantum dot-based electroluminescent light-emitting diodes

Fei Chen,^{†a} Qingli Lin,^{†b} Huaibin Shen^{ib*} and Aiwei Tang^{ib*}

Quantum dot light-emitting diodes (QD-LEDs) have been considered as one of the promising candidates for light sources in display applications owing to their good color saturation, high brightness, spectral tunability, and low processing cost. Thus far, the performance of red and green QD-LEDs has been significantly improved by optimizing QD materials and device architectures and by gaining a further understanding of their luminescence mechanism. However, as one of the three primary color sources, the performance of blue QD-LEDs still lags behind that of red and green QD-LEDs. Herein, we review the recent research progress in blue QD-LEDs, including Cd-based QD-LEDs, perovskite-based LEDs (PeLEDs), and heavy-metal-free QD-LEDs. The advances in Cd-based blue QD-LEDs are firstly introduced, and the efficiency has been improved from less than 0.1% to ~20% through the optimization of the components and surface ligands of QDs as well as the optimization of device architecture and interfacial engineering, and the aging mechanism of blue QD-LEDs is also discussed. Next, we review PeLEDs based on different types of perovskite materials, including two-dimensional (2D)/quasi-2D perovskites included in organic–inorganic halide perovskites and all inorganic perovskites. Currently, state-of-the-art PeLEDs fabricated from quasi-2D organic–inorganic perovskites exhibit the maximum external quantum efficiency (η_{EQE}) exceeding 11%. Furthermore, we summarize the research progress in heavy-metal-free blue QD-LEDs, which are fabricated from ZnSe, InP, Cu-based QDs and carbon dots. Although there is a huge gap in the device performance of heavy-metal-free LEDs as compared to the other two types of blue LEDs, the heavy-metal-free blue LEDs stand for a promising achievement toward their practical application. Finally, some challenges and possible development directions of blue QD-LEDs are suggested, which may speed up the further progress.

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1. Introduction

Typically, semiconductor quantum dots (QDs) contain a few hundreds to tens of thousands of atoms and the diameter of the QDs is ~1–20 nm, which is smaller than the Bohr exciton radius (the distance between an electron and a hole).¹ When the QD size is scaled to nanometers, the electronic energy level near the Fermi level splits from continuous into discrete energy levels, rendering the electronic structure, and optical and magnetic properties of materials tunable by varying the QD size.² Colloidal QDs have been a subject of active research in applications including solar cells,^{3,4} biological applications,⁵ photodetectors,^{6,7} and light-emitting diodes (LEDs)^{8–11} owing to their unique physical properties of size-dependent energy

band gap, narrow spectral emission bandwidths, broad spectral photoresponse from ultraviolet to infrared, and compatibility with the solution process. In particular, QD-LEDs are a promising candidate for next-generation displays owing to their high color purity and color-tunable emission achieved by changing the size of the core. Thus far, red QD-LEDs with a maximum external quantum efficiency (η_{EQE}) of 20.5% and a device lifetime of more than 100 000 h at 100 cd m^{−2} have been reported.¹² In 2019, by introducing compositional graded CdZnSe/ZnSe/ZnSeS/ZnS QDs with high photoluminescence (PL) quantum yields (QYs), suppressed nonradiative Förster resonant energy transfer and Auger recombination, and favorable valence band alignment for enhanced hole injection, Shen *et al.* demonstrated a hero η_{EQE} and current efficiency of 23.9% and 100.5 cd A^{−1}, respectively, for green QD-LEDs.¹³ Using the interconnecting layer (ICL) proposed therein, all-solution-processed tandem green and red QD-LEDs were successfully demonstrated with an impressive η_{EQE} of over 42% and 34%,^{14,15} respectively, which are the highest values ever reported and are comparable with those of state-of-the-art phosphorescent organic LEDs (OLEDs). Notably, the η_{EQE} of blue QD-LEDs has also increased from ~0.1% to

^a Key Laboratory of Luminescence and Optical Information, Ministry of Education, School of Science, Beijing Jiaotong University, Beijing 100044, China.
E-mail: awtang@bjtu.edu.cn

^b Key Laboratory for Special Functional Materials of Ministry of Education, Henan University, Kaifeng 475004, China. E-mail: shenhuaibin@henu.edu.cn

[†] These authors contributed equally.

19.8% because of their unique QD structure (*i.e.*, the composition gradient in the thick shells), the stable surface modification of the QDs (*i.e.*, appropriate surface ligands), the improved balance between hole and electron current (achieved using appropriate device architecture), and a deep understanding of the aging mechanism in blue QD-LEDs.^{16–18} However, the operational lifetime of these blue QD-LEDs is lower than that of the red and green QD-LEDs.^{19–21}

Besides Cd-based QDs, perovskite nanocrystals (NCs) have also attracted scientific attention owing to their narrow full-width half maximum (FWHM) and high PL QYs. Currently, green perovskite-based LEDs (PeLEDs) exhibit a record η_{EQE} that surpasses 20% by mixing a presynthesized CsPbBr₃ perovskite with a CH₃NH₃Br (MABr) additive. This enhanced device performance can be attributed to the formation of a CsPbBr₃/MABr quasi-core/shell structure that passivates the nonradiative defects present in CsPbBr₃ crystals, boosting the PL quantum efficiency.²² The η_{EQE} of red PeLEDs has also exceeded 20% (21.3%) through anion exchange from pristine CsPbBr₃ using halide-anion-containing alkyl ammonium and aryl ammonium salts.²³ Unfortunately, the η_{EQE} of blue PeLEDs is still less than 12%. In the case of both Cd-based QDs and perovskite NCs, intrinsic toxicity limits further commercialization owing to the release of free Cd²⁺ or Pb²⁺ ions into the environment. This limitation paved the way for introducing heavy-metal-free QDs, wherein a maximum luminance of 4490 cd m⁻² and a peak η_{EQE} of 7.8% of red QD-LEDs were reported using 1-dodecanethiol (DT) coated CIS/ZnS core/shell QDs as the emitting layer.²⁴ In addition, for InP-based QDs, the peak η_{EQE} of red and green QD-LEDs also exceeded 12% and 13%, respectively, through the optimization of the composition gradient in QD shells and the doping of the ZnO electron transport layer (ETL).^{25,26} Nevertheless, heavy-metal-free blue QD-LEDs have been rarely reported.

The poor performance of blue QD-LEDs may be attributed to the following phenomena: (1) low PL QYs of the blue QD film. Although QDs with high PL QYs have been synthesized, the PL QY of QD films has rapidly reduced owing to the energy transfer between QDs post purification and film formation. The low PL QY of the QD film results in a low recombination efficiency of electrons and holes in the emitting layer, leading to the low efficiency of the QD-LED. (2) Low charge-transfer efficiency between QDs in solid-state films. Typically, QDs include inorganic core/shell and surface organic ligands, which are necessary to determine the electrical properties of the QD layer. In general, previously, fatty acids with long hydrocarbon chains were chosen as surface ligands of colloidal QDs. Although these long-chain organic molecules can prevent QD aggregation in organic solvents and reduce QD surface defects, they also affect charge injection into QDs and the charge carrier transport within the QD layer by forming a dense insulating layer.²⁷ Hence, selecting appropriate ligands that can not only stabilize the QDs and maintain their intrinsic good photoelectric properties but also realize efficient charge transfer between QDs in solid-state films is necessary. (3) The imbalance between the electron and hole injection in blue QD-LEDs. In the conventional organic–inorganic hybrid QD-LEDs, a large interface barrier exists between the QD

emitting layer and the organic hole transport layer (HTL) formed by the deep valence band energy level for blue emission.²⁸ However, research on component and shell thickness of blue QDs and appropriate choice of charge transport materials is currently insufficient. (4) Charge transfers across the QD–ZnO junction and charge accumulation in the ETL generate positively charged blue QDs and negatively charged ZnO, reducing the efficiency of radiative recombination in the QD emitting layer.²⁹ (5) The human eye is insensitive to blue light. Compared with 530 nm-green light and 620 nm-red light, only 1/9 and 1/4 of the radiated power, respectively, of blue light is needed to achieve the same visual effect.

Therefore, the device performance of blue QD-LEDs can be substantially improved by overcoming the following challenges: the low quality of blue QD-emitting materials and the inappropriate hole transport materials and device structure for blue emission. Bearing this in mind, the performance of blue QD-LEDs can be optimized by designing nanostructures with low defects, appropriate band-gap levels, and high PL QYs; modifying the surface of semiconductor QDs with appropriate ligands; and optimizing the interfaces and device architecture of blue devices. In this review, we aim to present recent advances in the improvement of blue device efficiency, including Cd-based blue QD-LEDs, blue PeLEDs, and heavy-metal-free QD-LEDs. We firmly believe that the performance of blue QD-LEDs will considerably improve in the near future and become comparable with that of their red and green counterparts.

2. Cd-based blue QD-LEDs

Although few blue QD-LEDs with a maximum brightness of ~125 cd m⁻² and a FWHM $\geq$ 32 nm have been demonstrated, their emission spectra inevitably exhibited significant tails extending into the green and red regions of the spectrum.⁸ Until 2007, bright and color-saturated emission from blue LEDs was reported by Tan,³⁰ who structurally engineered core/shell CdS/ZnS QDs using low shell-synthesis temperature (~200 °C) (Fig. 1a). As shown in Fig. 1b and c, the device based on the core/shell QDs emitted pure blue light at 460 nm with a narrow FWHM of 18 nm, and the peak luminance reached up to 1600 cd m⁻². In particular, the long-wavelength tail of electroluminescence (EL) was minimized to less than 5% of the total emission. However, a large lattice mismatch of about 12% between CdS and ZnS causes interfacial strain, reducing the PL QY of the QDs. Therefore, Anikeeva *et al.* chose ZnCdS-alloyed QD cores as blue-emitting QDs with the PL peak at 460 nm and a high PL QY of 50%. Selecting spiro-TPD as the HTL and TPBi as the ETL, a peak η_{EQE} of 0.4% was obtained.³¹ TPBi, a small molecule deposited *via* vacuum deposition, has inferior thermal stability and is highly susceptible to degradation induced by oxygen and/or moisture. Hence, a solution-processable inorganic charge transport layer (CTL) is more desirable for stable and high-performance QD-LEDs. In 2011, there was a major breakthrough in the performance of QD-LEDs. Qian *et al.* synthesized crystalline ZnO nanoparticles (NPs) *via* a solution-precipitation

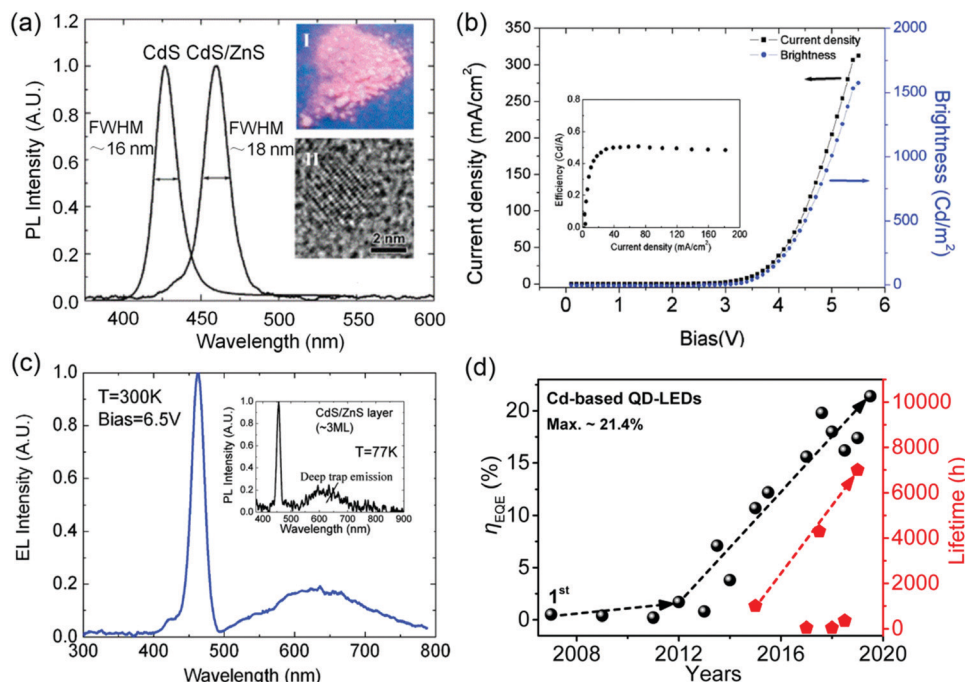


Fig. 1 (a) Photoluminescence (PL) spectra of CdS cores and core/shell CdS/ZnS QDs stored in tetrachloroethylene solvent. Inset I: Fluorescent image of QD power under UV light. Inset II: High-resolution transmission electron microscopy (HR-TEM) image of a core/shell CdS/ZnS QD. (b) J - L - V characteristics of a blue quantum dot light-emitting diode (QD-LED); the inset plots current efficiency vs. current density. (c) Electroluminescence (EL) spectrum of a QD-LED measured at 6.5 V. Inset: PL spectrum of CdS/ZnS QDs in the LED active region measured at 77 K. Reproduced with permission.³⁰ Copyright 2007, American Chemical Society. (d) Progress of blue QD-LEDs in the fraction of η_{EQE} and operating lifetime at an initial luminance of 100 cd m^{-2} .

method; these NPs adopted a wurtzite structure with an average diameter of 3 nm. Using these ZnO NPs as the ETL, all-solution-processable multilayer QD-LEDs were fabricated, in which all other layers were sequentially deposited on ITO *via* spin-coating except for the Al cathode, which was deposited *via* vacuum thermal evaporation. The resultant devices demonstrated a maximum brightness of 4200 cd m^{-2} for blue light, which was the highest brightness value reported at that time. Notably, that device exhibited a rather low turn-on voltage of 2.4 V, which was lower than the photon voltages due to a ZnO NP-induced Auger upconversion mechanism.³² As observed in Fig. 1d, since then, the η_{EQE} and lifetime of blue QD-LEDs have increased up to 19.8% and more than 7000 h, respectively, owing to the nanostructure engineering and stable surface modification of the QD emitting layer, the optimization of related device architecture, and deep understanding of the aging mechanism in blue QD-LEDs.^{19,33} In this section, we will review the recent progress in Cd-based blue QD-LEDs, divided into three aspects based on various improvements. In addition, Table 1 summarizes the detailed device characteristics of blue Cd-based QD-LEDs.

2.1 Nanostructure engineering and stable surface modification of the QD emitting layer

A prerequisite for realizing high-performance QD-LEDs is to use a QD film having a high PL QY. Although blue-emitting QDs with high PL QYs have been synthesized, the PL QYs of such QD films rapidly decrease owing to the energy transfer between QDs post purification and film formation. Therefore, introducing

intermediate shells, a composition-gradient in alloyed core/shell structures, and increasing the shell thickness are required for the blue QD film to maintain its initial PL QY. For example, Shen *et al.* prepared high-quality blue-green-emitting quaternary ZnCdSSe/ZnS QDs by adjusting the ratios between Zn and Cd and Se and S in the core. The PL QYs of as-synthesized ZnCdSSe/ZnS core/shell QDs can reach up to 50–75% with emissions between 450 and 550 nm. However, the η_{EQE} of the related blue QD-LEDs was only 0.8%, which may be due to the low PL QY of the blue-emitting QDs.³⁴ Lee *et al.* first reported highly fluorescent blue (452 nm) alloyed QDs of ZnCdS core/ZnS shell with a surprisingly high PL QY of 98% by optimizing the ZnS shelling time. Furthermore, although the 3 h-shelled ZnCdS/ZnS film exhibited a somewhat reduced PL QY of 86%, PL spectra between the solid-state QD film and dilute QD solution nearly matched without any noticeable peak shift. The exceptional performance of ZnCdS/ZnS QDs can be attributed to the following reasons. First, the long shelling duration could produce better surface passivation of ZnCdS core QDs with the ZnS shell, structurally improving the ZnCdS/ZnS interface *via* effective thermal annealing. Second, the shelling temperature of 310 °C, which is high enough to induce interfacial alloying, would render the core/shell boundary less abrupt, creating a relatively smooth interface in terms of composition. To verify the effectiveness of QDs with a thick shell, they fabricated QD-LEDs (Fig. 2a) by sequentially spin-coating the solutions of PEDOT:PSS and PVK in chlorobenzene, ZnCdS/ZnS QDs in hexane, and ZnO NPs in ethanol; consequently, an excellent η_{EQE} of 7.1% and a high luminance of 2624 cd m^{-2}

Table 1 Recent advances of blue Cd-based QD-LEDs

Year	Device structure	T_{50} (h)	PL peak (nm)	η_{EQE} (%)	η_A (cd A^{-1})	V_{on} (V)	L_{max} (cd m^{-2})	Ref.
2007	ITO/PEDOT:PSS/poly-TPD/CdS/ZnS/Al	—	460	—	0.506	2.5	1600	30
2009	ITO/PEDOT:PSS/spiro-TPD/ZnCdS/ZnS/TPBi/Al	—	460	0.4	—	—	—	31
2011	ITO/PEDOT:PSS/poly-TPD/CdSe/ZnS/ZnO/Al	—	470	0.22	0.32	2.4	4200	32
2012	ITO/ZnO/CdZnS@ZnS/CBP/MoO ₃ /Al	—	433	1.7	0.4	3.0	2250	50
2013	ITO/PEDOT:PSS/TFB/Zn _x Cd _{1-x} S _y Se _{1-y} /ZnS/ZnO/Al	—	460	0.8	0.52	2.4	4700	34
2013	ITO/PEDOT:PSS/PVK/ZnCdS/ZnS/ZnO/Al	—	452	7.1	2.2	4.0	2624	35
2014	ITO/PEDOT:PSS/TFB/Zn _x Cd _{1-x} S/ZnS/ZnO/Al	—	445	3.8	1.13	3.8	4100	36
2014	ITO/AZO:Cs ₂ CO ₃ /CdS/ZnS/poly-TPD/PEDOT:PSS/Al	—	—	—	0.06	3.6	249	52
2015	ITO/PEDOT:PSS/PVK/Cd _{1-x} Zn _x S@ZnS/ZnO/Al	1000	455	10.7	4.4	2.6	4000	20
2015	ITO/PEDOT:PSS/TFB/ZnCdS/ZnS-OT/ZnO/Al	—	441	12.2	1.9	2.6	7600	38
2017	ITO/PEDOT:PSS/PVK/Zn _x Cd _{1-x} S/ZnS/ZnO/Al	—	445	15.6	1.71	5.8	4500	39
2017	ITO/PEDOT:PSS/PVK/CdSe/ZnS/ZnO/Al	47.5	466	19.8	14.1	5.1	4890	33
2017	PET/ITO-Ag-ITO/ZnO/CdSe/ZnS/mCP/MoO ₃ /Al	—	478	—	16.1	2.8	18 127	55
2018	ITO/ZnO/CdZnSe/ZnS/mCP/MoO ₃ /Al	—	455	3.23	1.08	3.1	—	37
2018	ITO/PEDOT:PSS/PVK/ZnCdS/Cd _x Zn _{1-x} S/ZnS/ZnO/Al	47.4	445	18.0	3.4	6.5	6768	40
2018	ITO/PEDOT:PSS/TFB/ZnCdSe/ZnS/ZnS/PMMA/ZnO/Al	355	472	16.2	11.8	2.5	14 100	41
2018	ITO/PEDOT:PSS/poly-TPD/DNA/ZnCdS/ZnS/ZnO/Al	—	462	5.65	2.3	3.2	16 655	43
2018	ITO/PEDOT:PSS/PVK:TPD/ZnCdS/ZnS/ZnO/Ag	—	457	8.62	6.90	3.1	10 824	44
2018	ITO/PEDOT:PSS/PVK/ZnCdSeS/ZnS/Al/ZnO/Al	—	467	4.0	2.7	3.8	—	48
2018	ITO/PEIE/ZnO/ZnCdS/ZnS/PVK/MoO _x /Al	—	458	5.5	2.5	4.2	2640	51
2018	ITO/ZnO/CdSe/ZnS/PVK/TFB/PMMA/Al	—	454	5.99	1.99	4.1	4140	53
2019	ITO/PEDOT:PSS/TFB/CdSe/ZnSe/ZnO/Al	7000	476	8.05	—	2.5	62 600	19
2019	ITO/PEDOT:PSS/TFB:TCTA/ZnCdS/ZnS/AZO:ZrAcac/Al	—	464	10.7	5.7	3.0	34 874	45
2019	ITO/PEDOT:PSS/PVK/CdSe/ZnS/LiF/ZnO/Al	—	465	9.8	7.9	3.3	36 970	46
2019	ITO/PEDOT:PSS/TFB/ZnCdS/ZnS/TBS-PBO/ZMO/Al	—	449	17.4	—	3.0	7992.5	47
2019	ITO/PEDOT:PSS/poly-TPD/ZnCdS/ZnS/ZnO/Al:Al ₂ O ₃	—	452	8.92	3.29	3.4	27 753	49
2019	ITO/PEDOT:PSS/PVK/CdZnSeS/ZnS/ZMO/Al/HATCN/MoO ₃ /PVK/CdZnSeS/ZnS/ZMO/Al	—	474	21.4	17.9	8.8	26 800	54

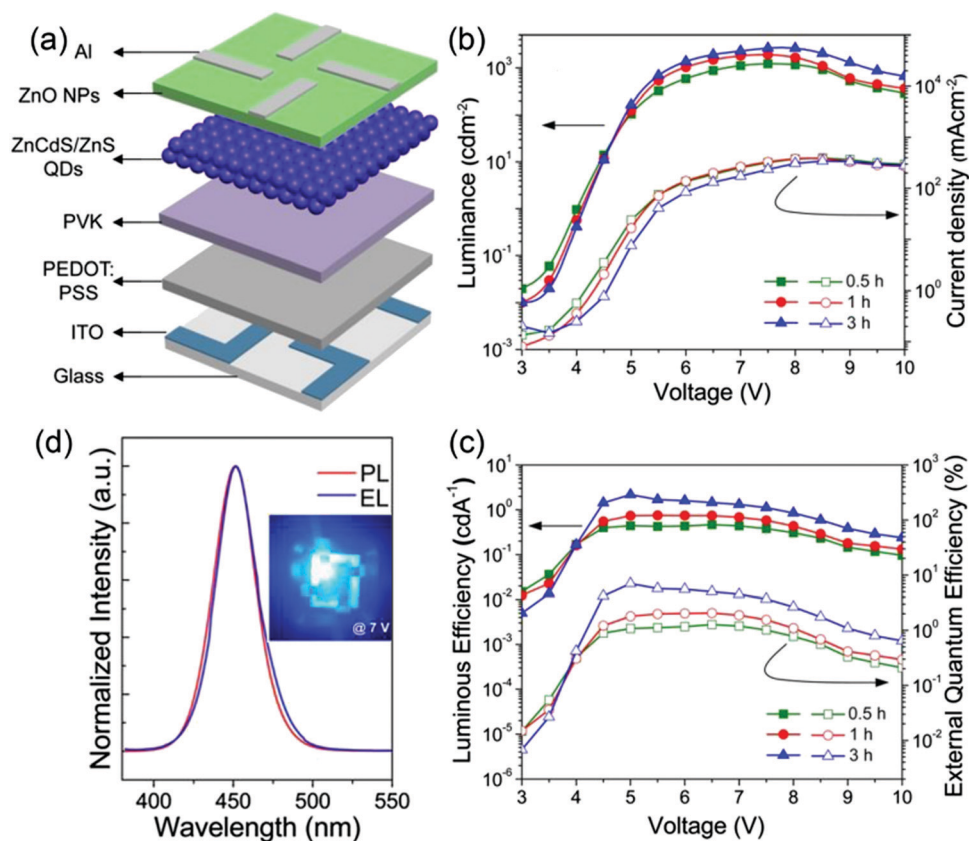


Fig. 2 (a) Schematic illustration of an all-solution-processed blue QD-LED with a multilayered structure, comprising ITO/PEDOT:PSS/PVK/ZnCdS/ZnS QDs/ZnO NPs/Al. Voltage-dependent variations of (b) luminance and current density and (c) current efficiency and η_{EQE} of QD-LED devices based on blue QDs with different shelling times. (d) Spectral comparison of the PL of the QD solution vs. the EL of the QD-LED. The inset shows the EL image at 7 V. Reproduced with permission.⁵⁵ Copyright 2013, American Chemical Society.

were obtained from 3 h-shelled QD-LEDs (as shown in Fig. 2b and c). In addition, as shown in Fig. 2d, the EL peak position (452 nm) and FWHM (31 nm) of the 3 h-shelled QD-LEDs almost matched with the PL of the dilute QD solution. However, the only disadvantage in this case was that the turn-on voltage was higher than 4 V.³⁵ Hence, to optimize ZnCdS/ZnS QD-based devices, Shen *et al.* reported a new method, called “nucleation at low temperature/shell growth at high temperature,” to synthesize high-quality violet-blue-emitting $\text{Zn}_x\text{Cd}_{1-x}\text{S}/\text{ZnS}$ core/shell QDs. The high growth temperature used therein not only promoted fast shell growth but also promoted the diffusion of S and Zn atoms into the $\text{Zn}_x\text{Cd}_{1-x}\text{S}$ core regions. Hence, the surface of $\text{Zn}_x\text{Cd}_{1-x}\text{S}$ cores could be restored and a consecutive change in lattice parameters could be achieved with $\text{Zn}_x\text{Cd}_{1-x}\text{S}$ buffer layers; additionally, stress defects were avoided. Furthermore, by selecting high-quality $\text{Zn}_x\text{Cd}_{1-x}\text{S}$ QDs having the highest PL QY as cores and designing a shell-thickness of ~ 9.5 monolayers, the PL QY of the core/shell QDs reached nearly 100%, which is the highest solution-based value for blue-emitting QDs reported at that time. More importantly, the new well-established protocols proposed by Shen *et al.* were easy to use on a large scale, and around 37 g of $\text{Zn}_x\text{Cd}_{1-x}\text{S}/\text{ZnS}$ core/shell QDs could be obtained in one batch reaction. The brightness and efficiency of QD-LEDs with a peak luminance of 4100 cd m^{-2} and a maximum η_{EQE} of 3.8% were demonstrated by utilizing $\text{Zn}_x\text{Cd}_{1-x}\text{S}/\text{ZnS}$ core/shell QDs as the emitting layer, simultaneously realizing high efficiency and high brightness at that time.³⁶ Despite these great advances, the turn-on voltage of more than 3 V and the operating voltage of more than 5 V are high, lowering the luminous power efficiency and reducing the operation lifetime of these devices. Additionally, lifetime test results have not been reported for any of the QD-LEDs. Hence, preparing high-quality QDs by carefully controlling their nanostructure and composition is necessary to obtain QD-LEDs with high performance. ZnS is predominantly used as the outer shell to form CdSe/ZnS or CdS/ZnS core/shell QDs. However, a large lattice mismatch between CdS or CdSe and ZnS will cause interfacial strain and reduce the PL QY. The gradual change in composition from the core to the outer shell can effectively relieve the interfacial strain between the core and the shell. In 2015, Qian *et al.* reported a graded, alloyed intermediate shell ($\text{Cd}_{1-x}\text{Zn}_x\text{Se}_{1-y}\text{S}_y$) sandwiched between the $\text{Cd}_{1-x}\text{Zn}_x\text{S}$ core and the Zn- and S-rich outer shell for blue QDs. This device exhibited a maximum η_{EQE} of 10.7% and a low sub-bandgap turn-on voltage of 2.6 V through careful optimization of the Cd- and S-rich or Zn- and Se-rich intermediate shell and the thickness of ZnS outer-shell-based QDs. Moreover, the lifetime of the device could reach up to 1000 h, marking the first time when the lifetime of a blue-emitting device was tested. The high device performance can be attributed to (1) the gradual change in composition from the core to the outer shell, which effectively relieved the strain caused by lattice mismatch between the core and the shell, thereby enhancing the QD PL efficiency, and (2) the thinner ZnS outer shell, which enhanced charge injection into QD-LEDs while maintaining high PL QY, generating high efficiency and long lifetime. The outstanding performance of blue QD-LEDs is a critical step toward the commercial realization of QD-LED display technology.²⁰

Typically, colloiddally synthesized QDs are capped with fatty acids having long hydrocarbon chains that reduce QD conductivity and increase the distance between QDs, affecting the charge injection/transport within the QD layer. Therefore, Cho *et al.* presented ligand-asymmetric Janus QDs to improve the device performance of QD-LEDs, in which the bottom ligand of QDs in contact with the ZnO ETL could effectively block electrons and the top ligand ensured uniform deposition of organic HTLs with enhanced hole-injection properties. Most importantly, Cho *et al.* obtained a peak η_{EQE} of 3.23%, evidencing two-fold enhancement as compared with QDs comprising oleic acid (OA) ligands (having a peak η_{EQE} of 1.49%). However, the record η_{EQE} achieved therein was still considerably lower than the calculated maximum device efficiency (est. $\eta_{\text{EQE}} = 8\%$) based on the film's PL QY (40%).³⁷ Therefore, further investigation of appropriate ligands will improve device performance. Short-chain thioalcohol ligands seem to be a good choice for the following reasons: their strong attachment to the QD surface, the effect of reduction in the space between QDs, the barrier thickness for hole injection into the QDs, and promotion of the valence band energy level of QDs. For instance, in 2015, Shen *et al.* replaced long-chain ligands (OA) with short ones (OT) to improve the carrier injection/transport properties of QDs. The short-chain ligands (OT) of QDs effectively enhanced hole injection and balanced the electron/hole current in devices. A low turn-on voltage of 2.6 V and a maximum η_{EQE} of 12.2% were obtained for these QD-LEDs with a PL peak of 443 nm.³⁸ Based on this work, the peak η_{EQE} of blue QD-LEDs was further improved by optimizing the ZnS shell thickness of $\text{Zn}_x\text{Cd}_{1-x}\text{S}/\text{ZnS}$ core/shell QDs and choosing appropriate hole transport materials. Although a device that used PVK as the HTL exhibited a peak η_{EQE} of 15.6%, its high turn-on voltage (5.80 V) resulted in a low maximum luminance of 4500 cd m^{-2} and a low power efficiency of 1.71 lm W^{-1} owing to the low hole mobility of PVK. However, even though the η_{EQE} of a device that used TFB as the HTL was only 11.9%, its maximum luminance reached up to 10400 cd m^{-2} with a turn-on voltage as low as 2.6 V. More importantly, the η_{EQE} of these thick-shell QD-based LEDs was relatively constant over a wide range of luminance values (from 10 to 4000 cd m^{-2}). In addition, these devices possessed long operating lifetimes with a T_{70} of 47 h at an initial luminance of 420 cd m^{-2} owing to the ability of the thick QD shells to more effectively confine excitons in the QD cores and minimize the impact on the exciton recombination characteristics resulting from potential variations in the chemical/electronic structures at QD surfaces during device operation such as oxidation/reduction, ligand attachment/detachment, and other defect formations.³⁹

Despite the above characteristics, the efficiency and device lifetime of blue QD-LEDs remain lower than those of green and red QD-LEDs, which have a maximum η_{EQE} exceeding 20% and a device lifetime exceeding 100 000 h.^{20,21} Hence, the efficiency and lifetime of deep blue QD-LEDs still need to be enhanced. Although the ZnS thick shell can suppress QD blinking and eliminate photobleaching, it can also create a natural band barrier for charge recombination in QDs. Therefore, based on their reported blue ZnCdS/ZnS core/shell QDs having a PL QY exceeding 90%, Shen *et al.* designed ZnCdS core/shell QDs with

uniform alloyed thick $\text{Cd}_x\text{Zn}_{1-x}\text{S}$ ($x = \text{constant}$) as the intermediate shell and ultrathin ZnS as the outermost shell. The $\text{Cd}_x\text{Zn}_{1-x}\text{S}$ intermediate shell not only confined excitons within the ZnCdS core to improve the PL QY but also lowered the injection barrier with ultrathin ZnS as the outermost shell. The resulting ZnCdS/ $\text{Cd}_x\text{Zn}_{1-x}\text{S}$ /ZnS core/shell QDs exhibited a high PL QY nearing 100% and a narrow FWHM ($<18 \text{ nm}$), ensuring high color purity of the related QD-LEDs. As shown in Fig. 3a and b, by changing the thickness of $\text{Cd}_x\text{Zn}_{1-x}\text{S}$ and ZnS and the ratio of Zn to Cd in the intermediate $\text{Cd}_x\text{Zn}_{1-x}\text{S}$ shell, the resulting device exhibited a maximum luminance of 6768 cd m^{-2} and a peak η_{EQE} of 18% based on ZnCdS/ $10\text{Cd}_x\text{Zn}_{1-x}\text{S}/2\text{ZnS}$ QDs. However, the turn-on voltage was a little high (up to $>6 \text{ V}$); hence, the lifetime of the QD-LEDs was only 47.4 h at an initial luminance of 100 cd m^{-2} .⁴⁰ Thus, improvement in the stability of blue QD-LEDs is imminent. In 2018, Lin *et al.* reported blue nonblinking ($>98\%$ “on” time) ZnCdSe/ZnS//ZnS QDs having a PL QY of 92% by choosing

ZnCdSe alloy QDs as a “good” core and an overcoat of the ZnS shell as the second shell (Fig. 3c). In addition, inserting PMMA between the QD and ZnO ETL interface to impede excessive electron flux produced a maximum luminance of $14\,100 \text{ cd m}^{-2}$ and a peak η_{EQE} of 16.2%, which can be maintained over 12% in the luminance range of $100\text{--}3000 \text{ cd m}^{-2}$ (as shown in Fig. 3d). Notably, as observed in Fig. 3e, a device lifetime of 355 h was obtained at an initial luminance of 100 cd m^{-2} . The enhanced device performance might originate from the reduced charge-injection imbalance obtained by inserting the PMMA interlayer, and it may have suppressed all nonradiative exciton quenching processes, such as nonradiative Auger recombination, dissociation of excitons under high fields, and electric-field-induced decrease in the PL efficiency of the QDs, due to the interfacial alloyed layer.⁴¹ However, to date, no blue devices have been reported that can simultaneously possess both high luminance and long lifetime. Recently, Shen *et al.* replaced the ZnS shell of ZnCdSe/ZnS core/shell QDs with a ZnSe shell,

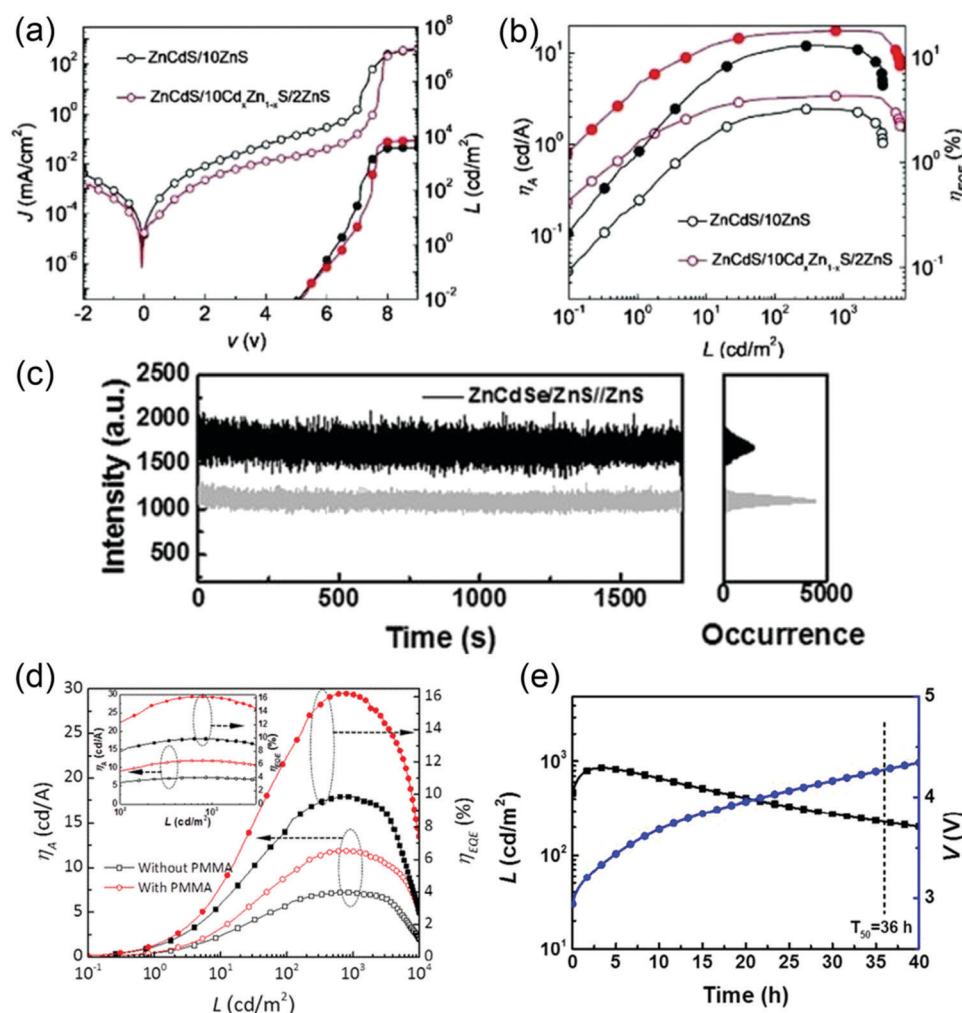


Fig. 3 (a) J – L – V characteristics and (b) current efficiency and η_{EQE} of blue QD-LEDs. Reproduced with permission.⁴⁰ Copyright 2018, Royal Society of Chemistry. (c) Right: Representative PL intensity time traces (1800 s) of ZnCdSe/ZnS//ZnS core/shell QDs under continuous wave (CW) laser excitation. Left: Photon counting histograms for the fluorescence traces shown on the right. (d) Current efficiency and η_{EQE} obtained using the luminance of QD-LEDs with and without a PMMA layer. The inset shows the corresponding efficiency in the luminance range of $100\text{--}3000 \text{ cd m}^{-2}$. (e) Operational lifetime characteristics of blue QD-LEDs. Reproduced with permission.⁴¹ Copyright 2018, American Chemical Society.

whose smallest valence band offset could match well with the HTL (TFB) compared with other commonly used materials, such as ZnS and CdS. When incorporating the resulting ZnCdSe/ZnSe core/shell QDs into all-solution QD-LEDs, the η_{EQE} was only 8.05%; however, the maximum brightness could reach up to $62\,600\text{ cd m}^{-2}$, which is the highest value reported to date. Moreover, the lifetime of blue QD-LEDs could reach as high as 7000 h at an initial luminance of 100 cd m^{-2} . This device meets the industrial requirements for lighting in terms of both brightness and efficiency (the minimum brightness and η_{EQE} requirements for lighting are 7000 cd m^{-2} and 6%, respectively) as well as device lifetime (for comparison, the average lifetime of incandescent lamps is around 1000 h). This work lays a foundation for lighting applications.¹⁹

2.2 Optimization of device architecture and interfaces between functional layers

Despite the significant progress, charge injection/transport imbalance still exists in devices. Apart from the optimization of the QD emitting layer, device architecture optimization also plays a vital role in the enhancement of device performance. In general, the device architecture can be classified into the traditional and inverted structures, in which the QD emitting layer is sandwiched between the ETL and HTL. Electrons and holes are transferred

from the cathode and anode to the QD emitting layer through the ETL and HTL, respectively. Therefore, interfacial engineering between the QD emitting layer and CTL together with the judicious choice of electron and hole transport materials will be crucial in balancing the carrier injection and charge transport and enhancing the radiative recombination of excitons in the QD emitting layer.⁴²

Owing to the large band-gap and low HOMO level of blue QDs and the relatively shallow HOMO of HTL materials, a large energy barrier exists between the QD emitting layer and HTL, causing difficulty in hole injection from the HTL to QDs. The imbalance between electron and hole injection causes excessive accumulation of electrons in the QD emitting layer, leading to nonradiative Auger recombination outside QDs. Additionally, superfluous electrons without recombination may leak into the HTL, resulting in parasitic emission, thereby affecting the device's color purity. To address this problem, many approaches have been proposed to achieve charge balance, such as amelioration of hole injection or reduction of electron injection. For example, Tan *et al.* reported a high-performance blue QD-LED by inserting a thin deoxyribonucleic acid (DNA) buffer layer between poly-TPD HTL and ZnCdS/ZnS core/shell QD emitting layer. The energy level diagram of the blue QD-LEDs in Fig. 4a demonstrates that the deep-lying HOMO level and the shallow LUMO level of DNA not

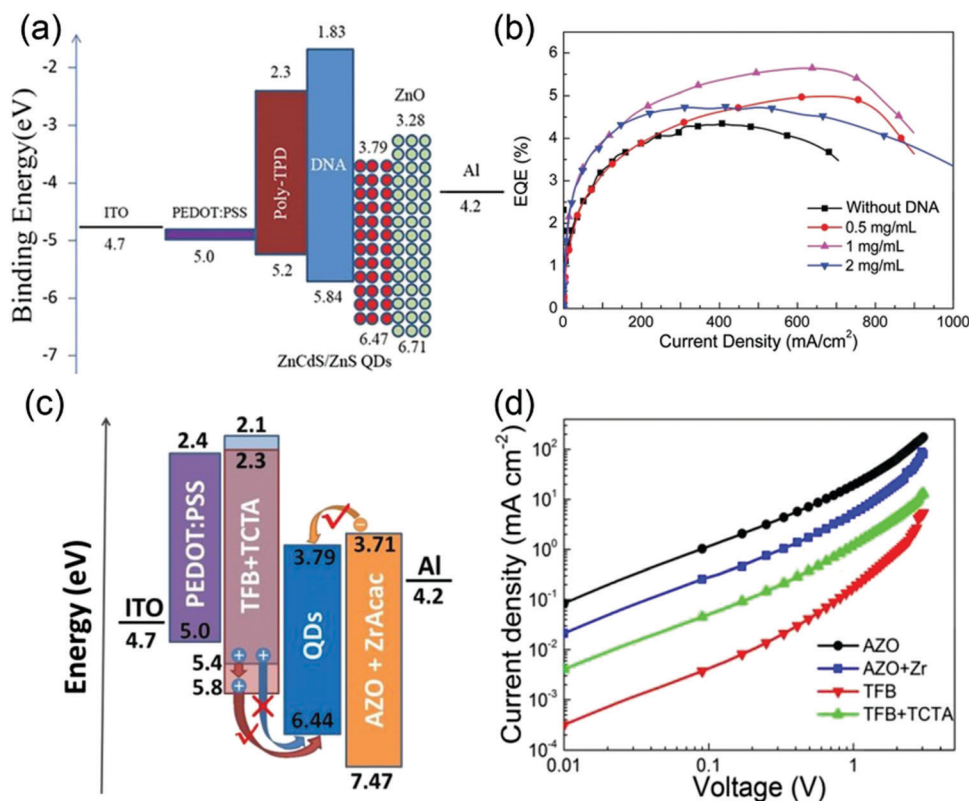


Fig. 4 (a) Energy level diagram of the blue QD-LEDs with a deoxyribose nucleic acid (DNA) modified HTL. (b) η_{EQE} –current density curves of devices with different DNA concentrations. Reproduced with permission.⁴³ Copyright 2018, John Wiley & Sons, Inc. (c) Energy level diagram of the blue QD-LED with TCTA-doped TFB as well as Al and ZrAcac-doped ZnO NPs as the HTL and ETL, respectively. (d) J – V characteristics of hole-only devices (ITO/PEDOT:PSS/HTL/QDs/MoO₃/Al) and electron-only devices (ITO/Al/QDs/ETL/Al). Reproduced with permission.⁴⁵ Copyright 2019, American Chemical Society.

only effectively enhance hole injection but also confine the injected electrons in the emitting layer, ensuring the charge balance in the QD layer. Consequently, after decorating with DNA, the maximum luminance and η_{EQE} significantly increased from 10 218 cd m^{-2} and 4.39% to 16 655 cd m^{-2} and 5.65%, respectively (Fig. 4b).⁴³ Apart from introducing additional functional layers, doping HTLs with high-mobility small molecules (such as TCTA or TPD) is another effective strategy to improve the charge mobility and reduce the hole injection barrier of polymer hole-transport materials. The dopant TPD was added to a PVK HTL to form a composite HTL. Because of the reduced injection barrier obtained using stepwise HOMO alignment as well as the reductions of recombination resistance and charge-transfer resistance assessed *via* electrochemical impedance spectroscopy (EIS) spectra, the maximum η_{EQE} and luminance were significantly increased from 2.37% and 3997 cd m^{-2} to 8.62% and 10 824 cd m^{-2} when the ratio of TPD and PVK was 1:1.⁴⁴ However, the low hole mobility ($10^{-6} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$) of PVK affected hole transport; hence, TFB (with a high hole mobility of $10^{-2} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$) is also a sensible choice for high-performance QD-LEDs. Unfortunately, its shallow HOMO produces a large energy offset between ZnCdS/ZnS QDs and TFB, rendering it difficult for hole injection. To solve this problem, Tan *et al.* doped TCTA with low HOMO into TFB as the HTL, guaranteeing both high hole mobility and smooth injection of holes. The corresponding energy level diagram is shown in Fig. 4c. Furthermore, Fig. 4d shows that the hole current density effectively increased in the hole device. To further enhance the balance of charge injection, Al and ZrAcac doped ZnO NPs were also chosen as the ETL, in which Al doping could adjust the bandgap of ZnO NPs and inhibit blinking in QDs, whereas ZrAcac doping could reduce the electron mobility of the ETL, boosting the charge balance in the QD emitting layer (Fig. 4d). Therefore, the resulting device based on TCTA-doped TFB as well as Al- and ZrAcac-doped ZnO NPs exhibited an exceptionally high luminance of 34 874 cd m^{-2} and η_{EQE} of 10.7%, which were 0.87 times and 1.30 times, respectively, higher than those of the control devices. These results prove that doping of CTLs is an effective and simple strategy to regulate charge balance in QD-LEDs.⁴⁵

In addition to the amelioration of hole injection, reduction of electron injection is also an effective method to promote the balance of charge injection, increasing the recombination efficiency in the QD layer. Thus far, various methods have been proposed to optimize electron injection, which could be divided into three categories: (1) inserting a blocking layer between the QDs and ZnO NPs. For instance, inserting a LiF interfacial layer between the ZnO ETL and the QD layer not only facilitated electron injection from ZnO to QDs through the electron tunneling effect, thereby reducing the accumulation of electrons at the ZnO/QD interface, but also suppressed exciton quenching in the QD layer caused by ZnO NPs with a high work function. Consequently, the blue QD-LEDs showed a maximum η_{EQE} of 9.8%, and the operational lifetime of the devices was improved two-fold owing to the more balanced charge injection.⁴⁶ However, the non-conductivity of LiF necessitates that its thickness should be precisely controlled, causing considerable difficulties in industrial production. Compared with

LiF insulating blocking layers, *tert*butyldimethylsilyl chloride-modified poly(*p*-phenylene benzobisoxazole) (TBS-PBO) has good conductivity, maintaining current density at a high level. Hence, a peak η_{EQE} of 17.4% and a maximum luminance of 7992.5 cd m^{-2} were obtained for the device when TBS-PBO was inserted between the ETL and QDs. In particular, a notable luminance of 4635 cd m^{-2} was achieved at the peak η_{EQE} of 17.4%, which was the highest luminance reported for blue QD-LEDs. This work offers a potential approach to achieve blue QD-LEDs with high luminance.⁴⁷ (2) Doping metallic ions to tune the conductivity and optical bandgap of ZnO NPs. According to Sun's report, through doping Al ions, the work function and conduction band edge of ZnO were raised by generating additional free electrons in the conduction band, effectively suppressing charge transfer at the QD/ZnO interface. Consequently, the blue QD-LEDs exhibited a maximum η_{EQE} of 4.0% and a decreased turn-on voltage from 4.1 to 3.8 V.⁴⁸ (3) Modifying the ZnO ETLs to tune the trap state density in ZnO NPs. For instance, post-annealing the ZnO film is the most typical treatment method that remarkably impacts device performance. Although the bandgap of the ZnO film is not influenced by annealing temperature, the gap state density decreases with increasing annealing temperature, indicating the drop in electron mobility in ZnO NPs. Additionally, they introduced Al:Al₂O₃ as the cathode to reduce the energy barrier between the QDs and the cathode, resulting in improvement of exciton confinement. Thus, with the good cooperation of the post-annealed ZnO ETL and the Al:Al₂O₃ cathode, a high brightness of 27 753 cd m^{-2} and η_{EQE} of 8.92% were obtained when the ZnO annealing temperature was 80 °C (Fig. 5a and b).⁴⁹ The electron mobility of the ZnO NPs can also be adjusted by controlling their particle size. Wang *et al.* synthesized two types of ZnO NPs A ($d = 3.626 \text{ nm}$) and B ($d = 3.888 \text{ nm}$) and investigated the effect of different-sized ZnO NPs on device performance. That study showed that small-size ZnO NPs A had a low surface roughness and more trap states than ZnO NPs B; hence, the electron-only device based on ZnO NPs B would have a larger leakage current than that based on ZnO NPs A. Consequently, the blue QD-LEDs based on ZnO NPs A and CdSe/ZnS QDs with a PL QY of 87% exhibited an exceptionally high η_{EQE} of 19.8% at a current density of 5.53 mA cm^{-2} , and the device efficiency is the highest value among previous reports (Fig. 5c). The QD-LEDs had pure blue light with a narrow FWHM of 20 nm corresponding to CIE 1931 color coordinates of (0.136, 0.078), which are very close to the NTSC 1953 standard of (0.14, 0.08) (Fig. 5d). Its color saturation property thus makes this QD-LED an ideal blue light source for display applications.³³

In the conventional QD-LED structure, an orthogonal solvent must be chosen to prevent physical damage during solution-based deposition. Now, the inverted device architecture is adopted to solve the problem of solvent selection restrictions. In addition, QD-LEDs with an inverted structure are highly advantageous for display applications because the bottom transparent cathode can be directly connected to n-channel metal-oxide or amorphous silicon-based thin-film transistor (TFT) backplanes. According to Kwak's report, the inverted blue device obtained by adopting the solution-processed ZnO NP film as the ETL and

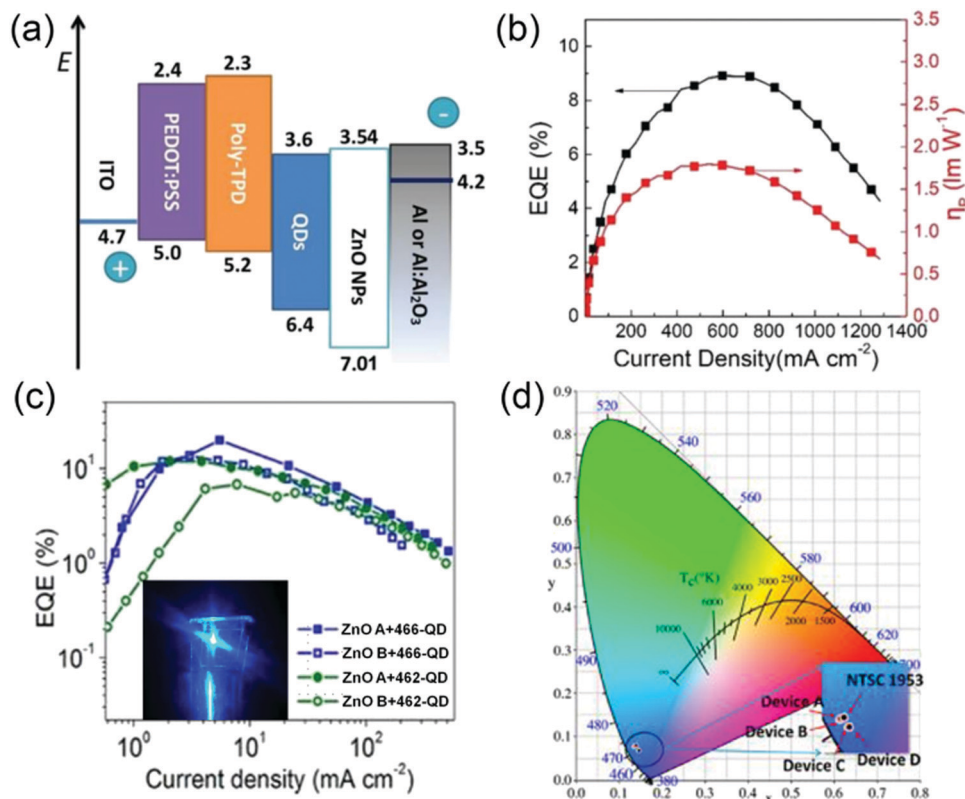


Fig. 5 (a) Energy level diagram of the as-prepared QD-LEDs. (b) η_{EQE} –power density curves of the best device with ZnO(a)/Al:Al₂O₃. Reproduced with permission.⁴⁹ Copyright 2019, John Wiley & Sons, Inc. (c) η_{EQE} –current density curves of devices with different ZnO and QDs. Inset: Image of the device with small-size ZnO NPs A and high PL QY (87%) QDs at a driving voltage of 7 V. (d) CIE properties. Reproduced with permission.³³ Copyright 2017, American Chemical Society.

vacuum evaporated CBP and MoO₃ as the HTL/hole injection layer (HIL) demonstrated a high efficiency of 1.7% and a low turn-on voltage of 3.0 V, which directly corresponded to the optical bandgap of blue Cd_{1-x}Zn_xS@ZnS QDs. This significantly improved performance might result from direct and efficient carrier recombination within QD layers caused by the facilitated charge carrier injection into QDs from ZnO or CBP.⁵⁰ Together with a deepening understanding of the physical mechanism of devices, the optimization of device structure and engineering of QD emitting material, the device performance of QD-LEDs has been further improved. For example, Zhong *et al.* presented an inverted blue QD-LED by inserting an ultra-thin PEI between the ITO and ZnO layers to decrease oxygen vacancy defects of ZnO and electron accumulation at the interface of the ZnO and QDs. Consequently, the blue QD-LEDs exhibited an increased peak η_{EQE} from 3.5% to 5.5%.⁵¹ However, vacuum evaporation is a high-cost technology that is unsuitable for large-area applications. Recently, all-solution-processed inverted blue QD-LEDs have been developed. The first one was reported by Castan *et al.*, who used polyoxyethylene tridecyl ether (PTE) to facilitate the deposition of aqueous PEDOT:PSS on a hydrophobic polymer hole transport material. However, the current efficiency of the blue QD-LEDs was merely 0.06 cd A⁻¹ owing to the effect of PTE on decreasing current flow through the device.⁵² To further resolve the solvent erosion of the HTL on the QD emitting layer, Liu *et al.* adopted the

polar solvent 1,4-dioxane as the solvent of the PVK HTL and phosphomolybdic acid hydrate (PMAH) dissolved in isopropanol as the HIL. The resulting blue QD-LEDs exhibited a maximum luminance of 1280 cd m⁻² and a peak η_{EQE} of 4.69%. To further enhance the EL performance of the devices, a double-layer HTL, PVK/TFB, was developed to provide stepwise hole injection, wherein the second HTL layer TFB dissolved in *p*-xylene prevented physical damage to the PVK layer. Therefore, a high η_{EQE} of 5.99% and a maximum brightness of 4140 cd m⁻² were obtained at an EL peak of 454 nm, and the device efficiency is the highest value among all inverted QD-LEDs, including thermally evaporated devices.⁵³ However, $\eta_{\text{EQE}} > 20\%$ has been routinely reported in red and green QD-LEDs, whereas it is still rarely reported in blue QD-LEDs. A promising approach to simultaneously achieve high efficiency and long operational lifetime is to employ a tandem structure, which has been widely adopted in OLEDs. In the case of blue QD-LEDs, an efficient tandem QD-LED with an η_{EQE} exceeding 21% was reported using an ICL with the structure of ZnMgO/Al/dipyrazino[2,3-*f*:2',3'-*h*]quinoxaline-2,3,6,7,10,11-hexacarbonitrile (HATCN)/MoO₃. The proposed ICL exhibits high transparency, efficient charge generation/injection capability, and high robustness to resist solvent damage during the deposition of the upper functional layers, and thus the highly bright blue QD-LEDs demonstrated a record η_{EQE} and current efficiency of 21.4% and 17.9 cd A⁻¹, respectively,

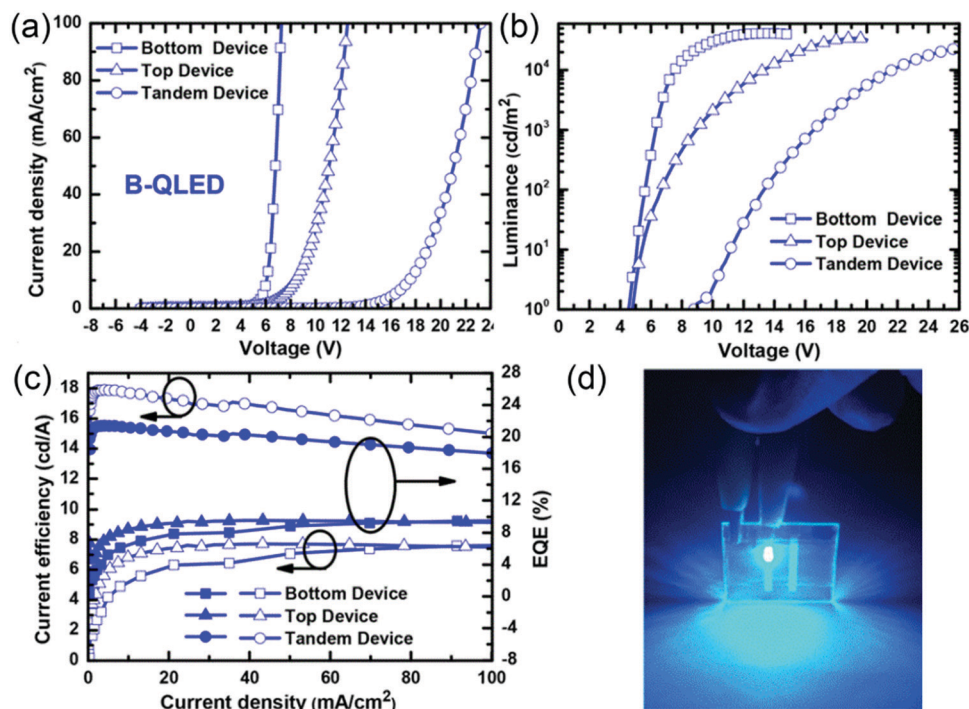


Fig. 6 (a) J - V characteristics, (b) L - V characteristics, and (c) current efficiency- J - η_{EQE} characteristics of the blue QD-LEDs. (d) Photos of the tandem blue QD-LEDs operated at a current density of 100 mA cm^{-2} . Reproduced with permission.⁵⁴ Copyright 2017, American Chemical Society.

and a high luminance of 26800 cd m^{-2} (Fig. 6); these results render blue QD-LEDs ideal candidates to bring QD-LEDs into the next generation of full-color displays and solid-state lighting markets.⁵⁴

All the results presented above are based on ITO-patterned glass substrates, whose rigidity limits the applications in foldable displays. Therefore, flexible photoelectric devices are necessary for next-generation displays. In 2017, Ji *et al.* reported high-performance red, green, and blue three-primary-color QD-LEDs based on the flexible PET (poly(ethylene-terephthalate))/ITO/Ag/ITO (IAI) transparent cathode for the first time. The IAI film on PET possesses similar transmittance to that of a pure ITO film, and ultrastability after bending 2000 times with a bending radius of 5 mm. When applied to QD-LEDs with a structure of PET/IAI/ZnO/QDs/mCP/MoO_x/Al, blue QD-LEDs display good EL performance with a maximum current efficiency (luminance) of 16.1 cd A^{-1} (18127 cd m^{-2}). This work unlocked a new opportunity for flexible QD-LEDs in terms of application to displays and lightings.⁵⁵

2.3 The aging mechanism of blue QD-LEDs

In the experiment, we find that the performance of QD-LEDs is significantly enhanced after shelf-aging for several days. This phenomenon can be termed as positive aging. Acharya *et al.* considered that positive aging was caused by the encapsulation resin, which slowly reacted with the ZnO surface and ambient carbon dioxide, thus forming zinc carbonate that helped to reduce ZnO NP defect densities and modified the ZnO/QDs or ZnO/Al interface.¹⁶ However, Chen *et al.* determined that the encapsulation resin did not play a major role in positive aging. To find out the exact origin of positive aging, the effects of

cathode materials, ETL materials, and QDs on the aging behavior were systematically investigated by Chen *et al.* To rule out the effect of the cathode, a device without an Al cathode was fabricated initially, and after five days, the Al cathode was deposited and the device was characterized immediately. As shown in Fig. 7a, the device with the Al cathode deposited after five days did not show positive aging and exhibited a decreased η_{EQE} from 5.9% to 3.7%, indicating that positive aging remained unaffected by QDs and ZnMgO ETL. However, the device with the Al cathode deposited initially exhibited an enhanced η_{EQE} from 5.9% to nearly 9.0% after five days, which might have been induced by the Al/ZnMgO interface. To precisely probe the Al/ZnMgO interface, the Al was removed *via* sputter etching and the surface was monitored *via* X-ray photoelectron spectroscopy (XPS) (Fig. 7b and c). The intensity of O-2s, which was related to the content of oxygen vacancies and oxygen bonds in the hydroxide, was higher than that of the fresh sample after five days of aging, whereas the Al-2s, corresponding to amorphous AlO_x, was also enhanced after five days. This can be attributed to the fact that the oxygen out-diffused from the ZnMgO layer and then reacted with Al, thus accumulating oxygen vacancies at the Al/ZnMgO interface and forming AlO_x. The oxygen vacancies at the Al/ZnMgO interface may have increased the electron concentration of ZnMgO and the AlO_x interlayer could have effectively suppressed the quenching of excitons through the metal electrode, enhancing the performance of blue QD-LEDs.¹⁷ Furthermore, positive aging can also be observed under accelerated processes, such as stressing by a continuous constant current and post metallization annealing (Fig. 7d and e).^{17,18}

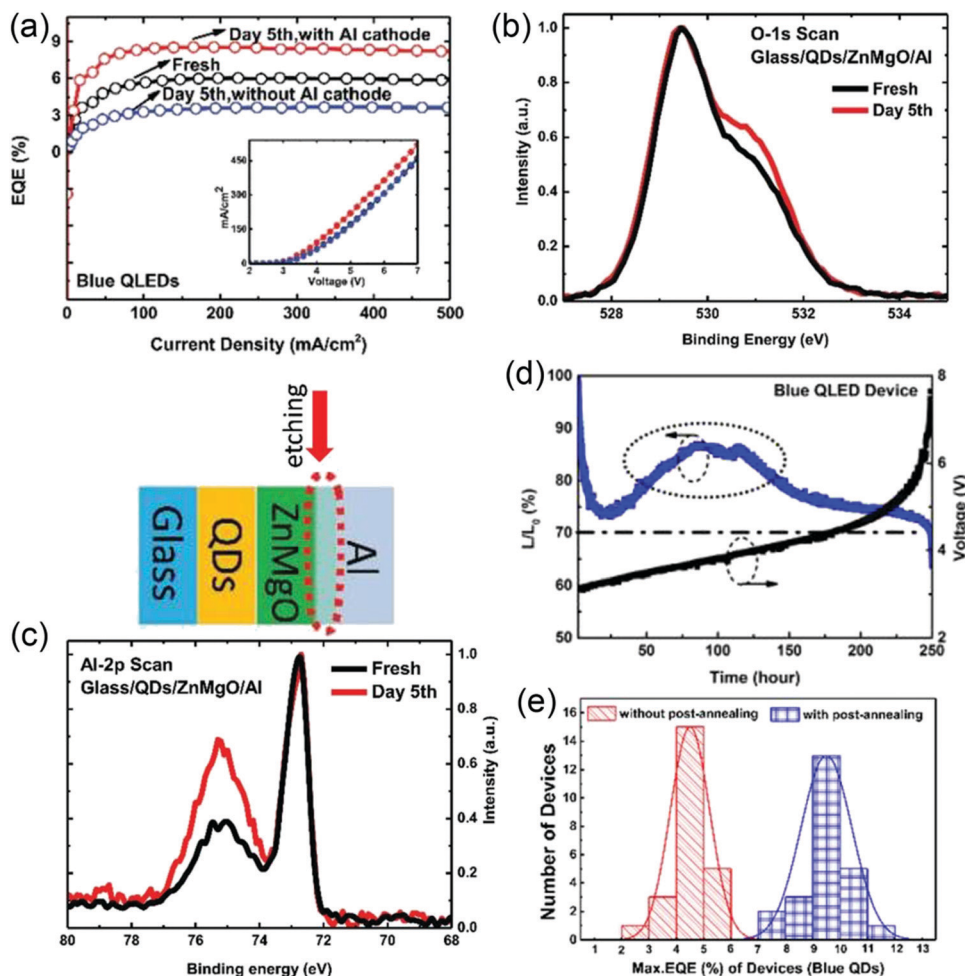


Fig. 7 (a) η_{EQE} -current density characteristics of blue QD-LEDs with and without the Al cathode placed initially. X-ray photoelectron spectroscopy (XPS) spectra of the (b) O 1s scan and (c) Al 2p scan at the Al/ZnMgO interface. (d) Lifetime of blue QD-LEDs with a conventional structure. Reproduced with permission.¹⁷ Copyright 2018, John Wiley & Sons, Inc. (e) Quantity statistics of the maximum η_{EQE} of blue QD-LEDs without and with post annealing. Reproduced with permission.¹⁸ Copyright 2018, American Chemical Society.

3. Blue perovskite LEDs

Owing to their bright tunable emissions and extremely narrow line widths, lead halide perovskite QDs have also attracted interest for potential optoelectronic device applications. In recent years, impulsive progress has been made in lead halide PeLEDs, and their maximum η_{EQE} has increased from 0.76% to more than 20%.^{22,23,56–64} In 2014, the first PeLEDs were reported by Tan *et al.*, and the resultant η_{EQE} values for near-infrared, red, and green emission at room temperature were 0.76%, 0.1%, and 0.018%, respectively. This demonstration of effective perovskite EL offers scope for developing this unique class of materials into efficient and color-tunable light emitters for low-cost display, lighting, and optical communication applications.⁵⁶ Ever since, the efficiency and luminance of PeLEDs have been drastically improved through composition and morphology engineering, surface modification, and device engineering. Thus far, near-infrared, red, and green PeLEDs have possessed high η_{EQE} values of up to 20.7%,⁶⁴ 21.3%,²³ and 20.3%,²² respectively. However, the development of blue PeLEDs is relatively slow. In

the following sections, we will review the recent progress of blue PeLEDs based on the types of perovskite materials, including 2D/quasi 2D perovskites included in organic-inorganic halide perovskites and all inorganic perovskites; additionally, Table 2 summarizes detailed recent advances in blue PeLEDs.

3.1 Organic-inorganic PeLEDs

The first blue PeLEDs based on organic-inorganic halide perovskites were reported by Kumawat *et al.* in 2015, who used PEDOT:PSS, MAPbBr_{1.08}Cl_{1.92}, and PCBM as the HIL, emitting layer, and ETL, respectively. However, the corresponding device exhibited a poor brightness of 2 cd m⁻² and a peak η_{EQE} of 0.0001% at room temperature owing to the pinholes in the perovskite film and emission quenching caused by PCBM.⁶⁵ Therefore, to reduce pinholes in the perovskite film, Kim *et al.* reported a triple-cation strategy that incorporated Cs cations into a mixed cation perovskite: MA_{0.17}FA_{0.83}Pb(Br_xI_{1-x})₃. In addition, using the two-step spin-coating method to form highly dense and pinhole-free perovskite films and by choosing an appropriate

Table 2 Recent advances in blue PeLEDs

QDs	Year	Device structure	PL peak (nm)	η_{EQE} (%)	η_{A} (cd A ⁻¹)	V_{on} (V)	L_{max} (cd m ⁻²)	Ref.
2D/quasi-2D perovskite included in organic-inorganic perovskite	2015	ITO/PEDOT:PSS/CH ₃ NH ₃ PbBr _{1.08} Cl _{1.92} /PC ₆₁ BM/Ag	471	<0.1	<0.1	—	2	65
	2017	ITO/ZnO/MA _{0.17} FA _{0.83} Pb(Br _x I _{1-x}) ₃ /α-NPD/MoO ₃ /Al	475	1.7	11.31	3.2	3567	66
	2019	ITO/PEDOT:PSS/PVK/CH ₃ NH ₃ PbBr ₃ /TPBi/CsF/Al	453	—	—	4.0	32	67
	2016	ITO/PEDOT:PSS/(PEA) ₂ PbBr ₄ /TPBi/Ca/Al	410	0.04	—	2.5	—	68
	2016	ITO/PEIE:ZnO/(4-PBA) ₂ PbBr ₄ :Cs/TFB/MoO _x /Al	491	0.015	—	2.9	186	69
	2016	ITO/PEDOT:PSS/PVK/OLA ₂ MA _{n-1} Pb _n Br _{3n+1} /TPBi/LiF/Al	456	0.0024	—	—	2	72
	2016	ITO/PEDOT:PSS/(PEOA) ₂ (MA) _{n-1} Pb _n Br _{3n+1} /TPBi/Ba/Al	460	1.1	2.1	—	19.5	74
	2017	ITO/PEDOT:PSS/(BA) ₂ (MA) _{n-1} Pb _n Br _{3n+1} /TmPyPB/CsF/Al	473	2.6	—	—	200	75
	2018	ITO/PEDOT:PSS/poly-TPD/CsPbBr ₃ NPs/TPBi/LiF/Al	488	0.1	—	—	25	70
	2018	ITO/PEDOT:PSS/NiO/PVK/PEA ₂ A _{n-1} Pb _n X _{3n+1} /TPBi/LiF/Al	477	1.5	2.8	—	2480	77
	2018	ITO/PEDOT:PSS/PVK/BA ₂ Cs _{n-1} Pb _n (Br/X) _{3n+1} /TPBi/Al	464	2.4	—	5.0	962	78
			486	6.2	—	4.5	3340	
	2019	ITO/NiO/TFB/PEA ₂ (RbCs) ₂ Pb ₃ Br ₁₀ /TmPyPB/LiF/Al	475	1.35	—	—	100.6	73
	2019	ITO/NiO _x /PSSNa/PA ₂ (CsPbBr ₃) _{n-1} PbBr ₄ /TPBi/LiF/Al	492	1.45	2.25	3.31	5737	76
	2019	ITO/PEDOT:PSS/PEA:CsPbCl _{0.9} Br _{2.1} /TPBi/LiF/Al	480	5.7	6.1	3.2	3780	79
	2019	ITO/NiO/TFB/PVK/CsFAPbBr ₃ /TPBi/LiF/Al	483	9.5	12.0	3.3	700	80
	2019	ITO/PEDOT:PSS/CsPbBr ₃ :PEACl:YCl ₃ /TPBi/LiF/Al	485	11.0	—	—	9040	81
All inorganic perovskite	2015	ITO/PEDOT:PSS/PVK/CsPb(Br/Cl) ₃ /TPBi/LiF/Al	442	0.07	0.14	5.1	742	82
	2016	ITO/PEDOT:PSS/PVK/CsPb(Br/Cl) ₃ -DDAB/TPBi/LiF/Al	490	1.9	—	3.0	35	85
	2017	ITO/NiO _x /CsPb(Br/Cl) ₃ /TPBi/LiF/Al	470	0.07	0.18	3.7	350	83
	2018	ITO/PEDOT:PSS/TFB/PFI/CsPb(Br/Cl) ₃ /TPBi/LiF/Al	469	0.5	—	—	111	84
			488	1.41	—	—	830	
	2018	ITO/PEDOT:PSS/TFB/PFI/CsPb(Br/Cl) ₃ :Mn/TPBi/LiF/Al	466	2.12	—	—	245	95

ETL of ZnO NPs and an HTL of α-NPD, the resulting device delivered a high brightness of 3567 cd m⁻², a peak η_{EQE} of 1.7%, and a low turn-on voltage of 3.2 V, which directly corresponded to the optical bandgap of the perovskite (Fig. 8a). The device also showed remarkably stable EL spectra at different bias voltages (Fig. 8b). However, the long-term stability of these devices under operating conditions remains a challenge. Kim *et al.* evaluated the operating lifetime of the device at room temperature and a constant current density for an initial luminance of 374.5 cd m⁻², the result of which demonstrated that during the first 25 min, the luminance increased to 948.5 cd m⁻² and then rapidly decreased to 88.5 cd m⁻² after 150 min.⁶⁶ In addition, to further promote their potential commercialization applications, efficient mass production of colloidal perovskite QDs has become an important research topic. In 2018, Zhong *et al.* reported a modified emulsion synthesis by introducing the phase transfer strategy to fabricate gram-scale CH₃NH₃PbBr₃,

which achieved effective extraction of newly formed perovskite QDs into non-polar solvent and avoided the degradation of perovskite QDs to a large extent. The as-fabricated CH₃NH₃PbBr₃ QDs exhibited a high PL QY of more than 70% and an emission peak of 453 nm with a FWHM of only 14 nm. However, the peak luminance of the related device was only 32 cd m⁻² and the FWHM of the EL spectrum was broadened to 16 nm.⁶⁷ Therefore, investigating more stable materials, such as 2D/quasi-2D organic-inorganic perovskite or all-inorganic perovskite materials, is necessary.

Although efficient blue electroluminescent three-dimensional (3D) PeLEDs have been successfully fabricated, the highest reported η_{EQE} for blue PeLEDs is still less than 5%. A convenient approach to create blue-emission perovskite films is transitioning from 3D lead halide perovskites into 2D or quasi-2D structures, which will introduce quantum confinement effects in these materials, and therefore, the PL peak can be easily tuned into blue or even violet wavelengths. For example, Liang *et al.* first

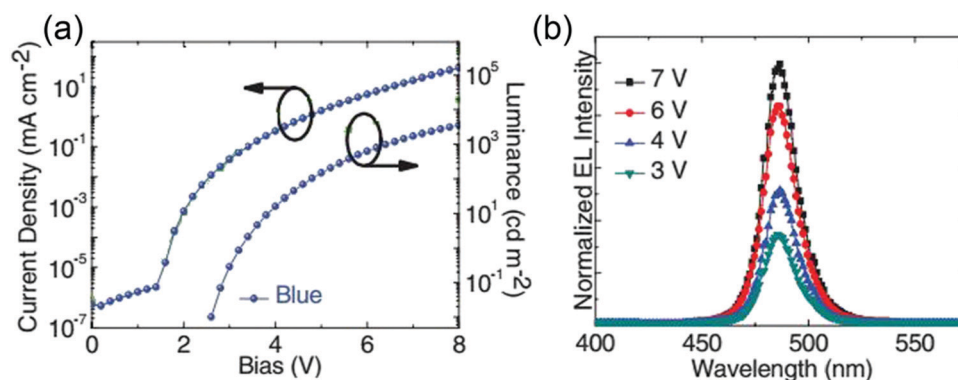


Fig. 8 (a) J - L - V characteristics for blue PeLEDs based on Cs-doped MA_{0.17}FA_{0.83}Pb(Br_xI_{1-x})₃. (b) EL spectra of the blue-colored PLEDs operating at various voltages. Reproduced with permission.⁶⁶ Copyright 2017, John Wiley & Sons, Inc.

reported violet PeLEDs based on 2D perovskites using (PEA)₂PbBr₄ (PEA = phenylethylammonium) NPs as the emitting layer and delivered a maximum η_{EQE} of 0.04% at room temperature.⁶⁸ By changing the organic spacer layer and mixing inorganic (Cs) cations, the related sky-blue PeLED based on ITO/ZnO/PEIE/(4-PBA)₂PbBr₄:Cs/TFB/MoO₃/Al exhibited a peak luminance of 186 cd m⁻² and a maximum η_{EQE} of 0.015% at an EL peak of 491 nm.⁶⁹ Moreover, Yang *et al.* demonstrated a simple one-pot colloidal approach to synthesize ultrathin CsPbBr₃ nanoplates (NPs) of tunable edge length and thickness. The blue PeLED of ITO/PEDOT:PSS/poly-TPD/CsPbBr₃ NPs/TPBi/LiF/Al exhibited a luminance of 25 cd m⁻² and an η_{EQE} of 0.1%.⁷⁰ However, 2D perovskites have limited PL QYs owing to their strong exciton-phonon coupling, causing difficulties in enhancing device performance.

The combination of 2D and 3D perovskites named quasi-2D perovskites with the formula of (RNH₃)₂A_{*n*-1}B_{*n*}X_{3*n*+1} (where *n* is the number of stacking perovskite octahedra and the bandgap can be tuned by changing the *n* values) was reported as a useful strategy to grant these materials with both compact pinhole-free morphology and excellent optoelectronic properties such as high PL QYs and good charge carrier balance.⁷¹ In 2016, Kumar *et al.* prepared colloidal, quantum-confined 2D perovskites OL₂MA_{*n*-1}Pb_{*n*}Br_{3*n*+1} using low-*k* organic host compounds to boost the exciton binding energy; the PL QYs of these perovskites could reach up to 40–90%. By tuning the *n* value from 1 to 10, they achieved tunable EL with λ = 432, 456, 492, and 520 nm at room temperature, and a blue PeLED with an emission peak at 456 nm exhibited an η_{EQE} of 0.0024% and a brightness of 2 cd m⁻².⁷² The poor device performance may be due to the inefficient internal energy transfer and Ostwald ripening because of the extremely low *n* (*n* ≤ 2) quasi-2D perovskites. So, Jiang *et al.* fabricated a Rb-Cs alloyed, $n_{\text{Rb0.6}}$ = 3 quasi-2D perovskite by tuning the 'A-site' cation composition, and demonstrated its great potential for further pure-blue LED application. The fabricated quasi-2D perovskite showed a prominent film PL QY of around 82% at 476 nm, and there was no PL spectral shift or broadening in these materials even after long-time thermal annealing. Eventually, the related PeLEDs exhibited a peak η_{EQE} of 1.35% and a lifetime of 14.5 min at the maximum η_{EQE} point. However, the maximum brightness was only 100.6 cd m⁻², which may be attributed to the poor perovskite film.⁷³ Chen *et al.* also found that device performance strongly depends on the perovskite film structure. They added a new organic material 2-phenoxyethylamine (POEA) into the CH₃NH₃PbBr₃ precursor solution, causing the transition from bulk film to layered structure through π - π stacking. This transition drastically changed the PL and EL of CH₃NH₃PbBr₃ with emission color being shifted from green to blue, and the associated performance of PeLEDs extraordinarily improved under optimized conditions, reaching a brightness of 19.5 cd m⁻² and an η_{EQE} of 1.1%.⁷⁴ In addition, the introduction of large-group ammonium halides is another method to obtain layered perovskites. Wang *et al.* found that the excess ethylammonium bromide (EABr) caused the phase transition from bulk to layered perovskite, accompanied by a blue shift of 40 nm. The associated PeLEDs with the EL located at 473 nm realized a peak η_{EQE} of 2.6% and a power efficiency of

1 lm W⁻¹ at 100 cd m⁻², exhibiting substantial improvement over those of previously reported analogous devices.⁷⁵ In addition to the improvements of perovskites, adopting a suitable carrier (hole and electron) transport/injection layer to reduce the carrier injection barrier is another way to improve the performance of blue PeLEDs. Although NiO_x was widely used as an efficient HTL in perovskite photovoltaics, it has been limitedly studied in blue PeLEDs because of its quenching effect on the perovskite emitting layer. Therefore, Ren *et al.* demonstrated an efficient hole transport bi-layer structure composed of PSSNa and NiO_x, which can not only inhibit the nonradiative decay between NiO_x and perovskite films by reducing NiO_x surface defects, but can also improve quasi-2D perovskite thin film quality by minimizing its pin-holes and reducing the film roughness. Eventually, the blue PeLED was achieved with a very low turn-on voltage of 3.31 V accompanied with an η_{EQE} of 1.45% and a remarkable luminance of 5737 cd m⁻².⁷⁶ However, the abovementioned blue PeLEDs usually exhibited poor color stability under LED operation because of the presence of multiple emission peaks arising from mixed phases. Therefore, Sargent *et al.* slowed the formation of the pure *n* = 1 phase to enhance the purity and monodispersity of *n* = 2, 3, and 4 phases by replacing long ligands (PEA) with shorter ones (iso-propylammonium, IPA). The as-synthesized perovskite films exhibited a single emission peak (474 nm) with a record PL QY of 88%, and the resulting PeLEDs exhibited a maximum luminance of 2480 cd m⁻² and a peak η_{EQE} of 1.5%. Particularly, the EL peak of 490 nm and the FWHM of 28 nm remained unchanged when the applied voltage increased from 5 to 10 V.⁷⁷ Besides, Vashishtha *et al.* adopted butylammonium as a separating ligand to create BA₂Cs_{*n*-1}Pb_{*n*}(Br/Cl)_{3*n*+1} 2D/3D mixed halide thin films, in which changing the butylammonium concentration could optimize the number of CsPbBr₃ monolayers. When they were incorporated into PeLEDs with a structure of ITO/PEDOT:PSS/PVK/BA₂Cs_{*n*-1}Pb_{*n*}(Br/X)_{3*n*+1}/TPBi/Al, a record efficiency of 6.2% at 487 nm for blue-emitting PeLEDs was obtained, and the peak luminance could reach up to 3340 cd m⁻² (Fig. 9a and b).⁷⁸ However, the relatively poor thermal stability is detrimental to device stability, which requires a high working voltage and hence a strong electric field for operation. Therefore, a much more stable perovskite system is required for blue PeLEDs. Among the perovskite family, an all-inorganic perovskite system possesses much better structural and thermal stability than an organic-inorganic hybrid system. Hence, a more stable blue PeLED can be obtained by forming a quasi-2D perovskite from a green-emissive all-inorganic 3D perovskite of CsPbBr₃. In 2019, Li *et al.* first synthesized CsPbCl_{0.9}Br_{2.1} with emission at 484 nm through composition engineering by incorporating Cl into the CsPbBr₃ lattice, and then introduced phenylethylammonium bromide (PEABr) into the CsPbCl_{0.9}Br_{2.1} perovskite to form a quasi-2D perovskite thin film, in which PEABr could efficiently passivate the traps in perovskite thin films, substantially improving their PL QYs from 0.15% to 27%. Consequently, a highly efficient blue PeLED with a maximum η_{EQE} of 5.7% and a peak luminance of 3780 cd m⁻² at an EL peak of 480 nm was successfully demonstrated. More importantly, the operation lifetime of the

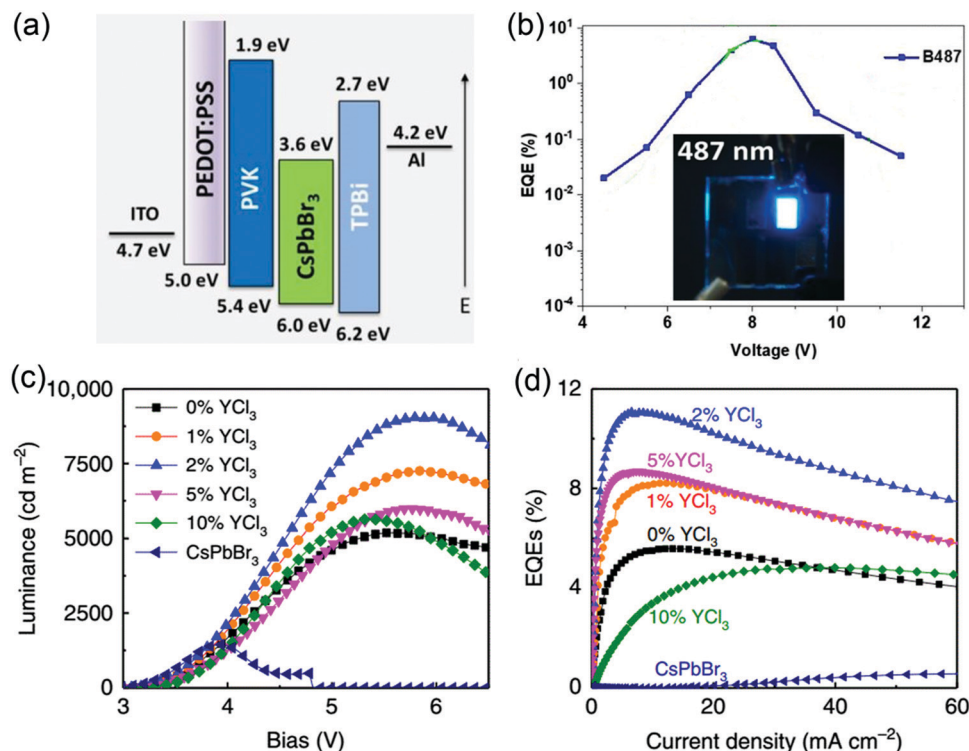


Fig. 9 (a) Energy diagrams for ITO/PEDOT:PSS/PVK/BA₂Cs_{*n*-1}Pb_{*n*}(Br/X)_{3*n*+1}/TPBi/Al LEDs. (b) η_{EQE} -voltage for B487 LEDs for champion devices. Inset: Photos of these LEDs. Reproduced with permission.⁷⁸ Copyright 2018, American Chemical Society. (c) Luminance-voltage and (d) η_{EQE} -current density curves of CsPbBr₃:PEACl (1:1) devices with different ratios of YCl₃. Reproduced with permission.⁸¹ Copyright 2019, Nature Publishing Group.

devices was studied under a constant applied voltage of 4.4 V, achieving a half lifetime of around 10 min, which is one of the best results among the reported lifetimes of blue-emitting perovskite LEDs. However, the EL spectrum of the device continuously shifted from 480 nm to 488 nm during the operating time of 9 min, which can be attributed to the possible phase separation and other degradation processes that require further investigation.⁷⁹ In addition, a considerable amount of excess cations of PBA⁺ is usually used to avoid over-growth of the all-bromide perovskite crystals; however, the insulating of PBABr can impede charge injection into the perovskite films and influence the energetic coupling between the quasi-2D phases and the quantum-confined nanoparticles. Hence, Liu *et al.* used ethyl acetate as an antisolvent, which dissolved PBABr and did not damage the perovskite layer or the polymer layers underneath, to deposit the perovskite layer from a precursor solution of PBABr, formamidinium bromide (FABr), CsBr, and PbBr₂ (1.1:0.3:0.7:1 molar ratio) dissolved in dimethyl sulfoxide. A symmetric peak centered at 483 nm with a narrow FWHM of 26 nm and a high PL QY of 60% was obtained, and high-performance PeLEDs with a record η_{EQE} of 9.5% and a half lifetime of 250 s at an initial brightness of 100 cd m⁻² were successfully demonstrated using these perovskites as emitting materials. This result showed the bright outlook of blue PeLEDs for lighting and display applications. However, the maximum luminance was only 700 cd m⁻².⁸⁰ To date, a record efficiency of 11.0% and a maximum luminance of 9040 cd m⁻² were obtained in sky-blue PeLEDs by blending phenylethylammonium chloride

(PEACl) into CsPbBr₃ to form a mixture of 2D and 3D perovskites, followed by the introduction of yttrium(III) chloride (YCl₃) into the mixture to further enhance the PL QY from 19.8% to 49.7% (Fig. 9c and d). The incorporation of YCl₃ served the following three functions: (1) reducing the nonradiative recombination centers in PEACl:CsPbBr₃ (1:1) films, induced by accumulating Y on the film surfaces or grain boundaries; (2) improving spectrum stability by drastically suppressing ion migration in these films; (3) increasing the bandgap of the grain shell, which confined the charge carriers inside grains for efficient radiative recombination. Furthermore, a blue PeLED with an η_{EQE} of 4.8% was also obtained by adding 10% YCl₃ in the perovskite film. This work provides a new method to improve PL QYs in Cl:Br-based perovskites and related PeLEDs.⁸¹

3.2 All-inorganic PeLEDs

Compared to hybrid organic-inorganic halides, all-inorganic perovskites exhibit higher stability and have huge potential in various optoelectronic applications. The first research attempt on blue all inorganic PeLEDs was made by Song *et al.*, who adopted the hot-injection method to synthesize high-quality blue CsPb(Br_{1-x}Cl_x)₃ QDs with a PL QY ranging between 60% and 90% and a narrow FWHM of 23 nm. The related PeLEDs based on PVK as the HTL and TPBi as the ETL exhibited a turn-on voltage of 5.1 V (at 1 cd m⁻²), a peak η_{EQE} of 0.07%, a maximum brightness of 742 cd m⁻², and bright blue emission ($\lambda = 455$ nm).⁸² Furthermore, in 2017, Yao *et al.* demonstrated the high brightness of blue all-inorganic PeLEDs with an emission peak of 470 nm and a

narrow FWHM by directly mixing CsPbBr₃ and CsPbCl₃ NCs dispersed in appropriate solvents. XRD, XPS, and UPS tests confirmed the production of a new-phase CsPbBr_xCl_{3-x}, where Br⁻ was partially replaced by Cl⁻. The resulting PeLEDs that utilized NiO_x as the HTL exhibited an enhanced current efficiency of 0.18 cd A⁻¹ and a maximum luminance of 350 cd m⁻².⁸³ However, severe emission quenching was observed at the NiO_x/CsPb(Br_xCl_{1-x})₃ interface, which could be suppressed using TFB/PFI (perfluorinated ionomer) instead of NiO_x. Such a modification can further improve the η_{EQE} to 0.5% at an emission wavelength of 469 nm.⁸⁴ However, the highest η_{EQE} reported is still less than 1%, which is partly due to the QDs capped with relatively insulating long ligands that are required for the processing and stability of the QDs. Maybe replacing these long ligands (usually oleylamine (OAm) and oleic acid (OA)) with shorter ligands can be an effective method to realize efficient blue PeLEDs. Pan *et al.* capped the synthesized CsPbBr_xCl_{3-x} NCs with didodecyl dimethyl ammonium bromide chloride (DDABC) *via* a ligand-exchange strategy that included an intermediate step to desorb protonated OAm and achieved a sky-blue PeLED with a turn-on voltage of 3.0 V and a maximum η_{EQE} of 1.9%. This work represented a major leap for perovskites and paved the way to further exploit PeLEDs in optoelectronics through judicious surface engineering.⁸⁵

The above reports on blue CsPbX₃ perovskite NCs were all realized by incorporating chlorine into bromine-based perovskites to tune the bandgap. Nevertheless, these mixed-halide perovskites exhibit intrinsic phase instability, particularly under light and electrical potential, which easily produces halide migration.⁸⁶ Consequently, the EL spectra of the corresponding PeLEDs spontaneously transform from blue to green within minutes under device operation.⁸⁷⁻⁸⁹ Thus, to prevent EL transformation caused by halide migration, the shape control of colloidal NCs and doping of NCs are used to adjust their optical and electronic properties. For example, Xu *et al.* developed a shape-controlled synthesis of blue-emission CsPbBr₃ perovskite NCs at a relatively low temperature (90 °C) by tuning the amounts of oleic acid (OA) and oleylamine (OM). When they directly added 0.6 mL of OA and 0.3 mL of OM into the reaction system, monodispersed, spherical QDs with a narrow PL emission peak at 453 nm were obtained. By adjusting the amounts of OA and OM from 0.3 mL of OA and 0.7 mL of OM and 0.5 mL of OA and 0.5 mL of OM to 0.8 mL of OA and 0.2 mL of OM, they acquired lamellar-structured 0D QDs and face-to-face stacking 2D nanoplatelets in addition to flat-lying 2D nanosheets with PL emissions at 472, 449, and 452 nm, respectively. This is different from the original green emission (~510 nm) of CsPbBr₃ nanocubes, which may be ascribed to the strong quantum confinement effects of these various nanoshapes. Such superior shape-controlled and PL-tunable merits made them favorable for optoelectronic devices, particularly for blue PeLEDs.⁹⁰ Furthermore, the doping of Al³⁺, Cu²⁺, La³⁺, and F⁻ ions as co-dopant can provide CsPbBr₃ NCs with a strong blue emission.⁹¹⁻⁹³ For instance, the doping of Cu²⁺ ions into the perovskite lattice to form CsPb_{1-x}Cu_xBr₃ QDs could not only improve their thermal stability but also enhance the PL QY from 23% to more than 80%.⁹² However, the doped blue NCs have not been applied to PeLEDs. Previous reports confirmed

that Mn-doping could significantly increase the PL QY of the perovskite film and enhance the stability of CsPbBr₃ NCs in green PeLEDs.⁹⁴ However, these Mn-doping sites could be efficient emitters in their own right. Hence, Congreve *et al.* adopted a two-step sequential synthesis approach to carefully tune the Mn-doping to obtain an emission peak of ~468 nm while preventing significant Mn emission. Fig. 10a and b show that Mn-doping could increase PL QYs from 9% for undoped films to 28% for the 0.19% doped sample. In addition, a significant increase in emissive state lifetime was observed after achieving 0.19% PL QY using the Mn dopant, indicating a reduction in trap-state densities. To identify the engineering benefits of this material improvement, Congreve *et al.* fabricated PeLEDs with PEDOT:PSS and TFB/PFI as the HIL/HTL together with TPBi as the ETL, and all samples exhibited pure blue emission only from the perovskite emitter layer (see Fig. 10c). In particular, up to four-fold improvement in η_{EQE} (reaching a maximum of 2.12%) was obtained with 0.19% Mn-doping, which, to the best of our knowledge, is the highest value at this wavelength range for all-inorganic PeLEDs (Fig. 10d). Mn-Doping provides a straightforward method to increase the performance of blue-emitting PeLEDs.⁹⁵

4. Heavy-metal-free QD-LEDs

The above described QD-LEDs mostly depend on Cd-based or Pb-based QDs, and their intrinsic toxicity limits their further commercialization because of strict environmental regulations such as the Restriction of Hazardous Substances (RoHS) directive in Europe. Hence, heavy-metal-free QD-LEDs are desirable in future displays and lighting. Thus far, several heavy-metal-free material systems have been developed, including ZnSe-based, InP-based, Cu-based and carbon dot (CD)-based QD-LEDs. The recent advances in heavy-metal-free blue QD-LEDs are summarized in Table 3.

4.1 ZnSe-based QD-LEDs

Owing to the large bandgap of 2.7 eV, ZnSe can be large sized to obtain relevant blue color wavelength. The first blue/violet QD-LED based on Cd-free ZnSe/ZnS QDs with the PL peak located at 420 nm was reported by Xiang *et al.*, delivering a reduced turn-on voltage of 3.5 V and a maximum η_{EQE} of 0.65% with PVK as the HTL. The achieved η_{EQE} of 0.65% suggested that Cd-free ZnSe/ZnS QD-LEDs would be a promising candidate for short-wavelength LEDs.⁹⁶ In the following year, Ji *et al.* demonstrated color-saturated deep-blue emission with a narrow FWHM of 15 nm at $\lambda = 441$ nm in an inverted structure, the maximum brightness and current efficiency of which could reach up to 1170 cd m⁻² and 0.51 cd A⁻¹, respectively. The poor device performance may be attributed to the low PL QYs and stability of QDs in the violet-blue area.⁹⁷ For improving the PL QY of ZnSe/ZnS core/shell QDs and the performance of the related QD-LEDs, Shen *et al.* adopted “a low temperature injection and high temperature growth” method together with the octanethiol ligand to synthesize nearly monodispersed

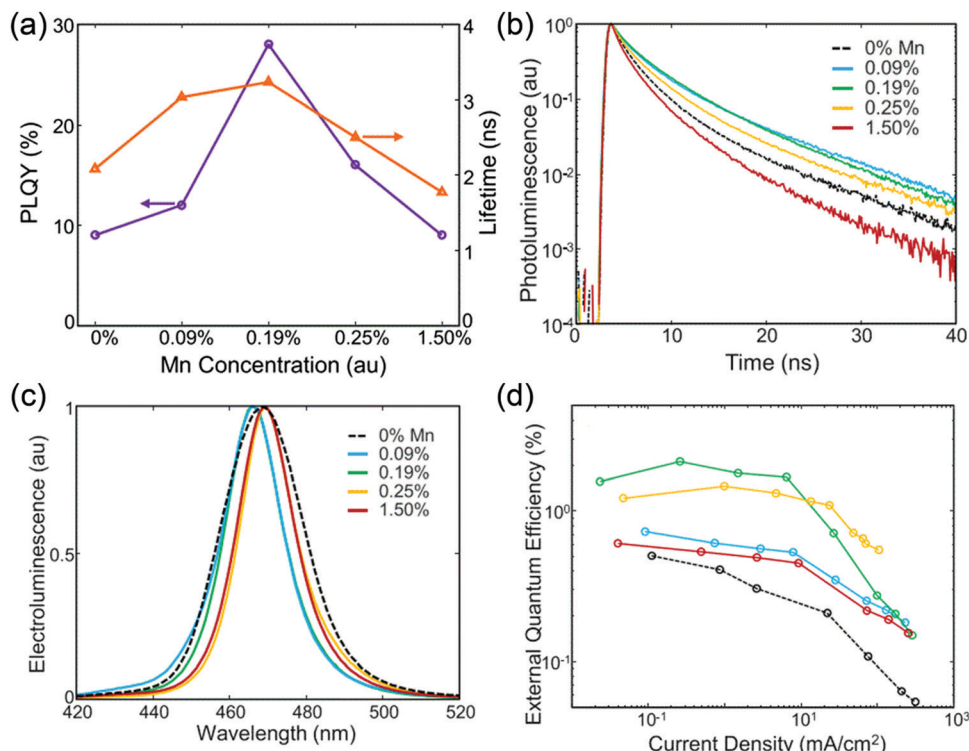


Fig. 10 (a) PL QY and 1/e lifetime of the perovskite emission of thin films of NCs. (b) Time-resolved perovskite luminescence of the different NC species. (c) EL spectra of the fabricated PeLEDs with NCs of varying Mn content. (d) η_{EQE} as a function of current density for the fabricated PeLEDs. Reproduced with permission.⁹⁵ Copyright 2018, Elsevier Ltd.

Table 3 Recent advances in heavy-metal-free blue QD-LEDs

QDs	Year	Device structure	PL peak (nm)	η_{EQE} (%)	η_{A} (cd A ⁻¹)	V_{on} (V)	L_{max} (cd m ⁻²)	Ref.
ZnSe-based QDs	2012	ITO/PEDOT:PSS/PVK/ZnSe/ZnS/ZnO/Al	420	0.65	—	—	—	96
	2013	ITO/ZnO/ZnSe/ZnS/TCTA/MoO ₃ /Al	441	—	0.51	4.0	1170	97
	2015	ITO/PEDOT:PSS/PVK/ZnSe/ZnS/ZnO/Al	425	7.83	1.38	7.0	2623	98
	2019	ITO/PEDOT:PSS/PVK/ZnSeTe/ZnSe/ZnS/ZnMgO/Al	441	4.2	2.4	4	1195	102
InP-based QDs	2017	ITO/ZMO/InP/ZnS//ZnS/CBP/MoO ₃ /Al	477	—	—	—	90	107
	2020	ITO/PEDOT:PSS/TFB/InP/GaP/ZnS//ZnS/ZnO/Al	488	1.01	—	2	3120	108
Cu-based QDs	2014	ITO/PEDOT:PSS/PVK/CGS/ZnO/Al	479	0.007	0.01	—	39	115
	2019	ITO/PEDOT:PSS/PVK/CZGS:Mn/ZnS/ZMO/Al	470	4.2	2.3	4.9	1352	117
	2019	ITO/PEDOT:PSS/PVK/ZCGS/ZnS/ZMO/Al	474	7.1	11.8	4.0	1404	118
Carbon dots	2013	ITO/PEDOT:PSS/poly-TPD/CDs/TPBi/LiF/Al	420	—	—	5	24	120
	2016	ITO/PEDOT:PSS/CDs/TPBi/Ca/Al	430	—	0.084	4.7	136	121
	2018	ITO/PEDOT:PSS/PVK:CDs/TPBi/LiF/Al	467	—	—	8.5	569.8	122
	2018	ITO/PEDOT:PSS/PVK:CDs/TPBi/Ca/Al	472	—	1.22	4.3	1882	123
	2019	ITO/PEDOT:PSS/TFB/PVK:CDs/TPBi/LiF/Al	433	4.0	2.6	—	5240	124

ZnSe/ZnS core/shell QDs with high PL QYs of more than 80%, high color purity with a FWHM of about 12–20 nm, and good color tunability in the violet-blue range from 400 to 455 nm achieved by adjusting the reaction time and temperature of the shell (Fig. 11a). Significantly, such ZnSe/ZnS core/shell QDs exhibited very good chemical/photochemical stability under ultraviolet (UV)-irradiation and repeated purification processes, which was essential for the high performance of QD-LEDs; thus, the highly bright violet QD-LEDs were successfully demonstrated with a fully solution-processed method, exhibiting a maximum

luminance of 2632 cd m⁻² and a peak η_{EQE} of 7.83% with the EL peak at 437 nm (Fig. 11b).⁹⁸ However, their EL wavelength (437 nm) was quite beyond the scope of the desired blue color (440–460 nm). To obtain ZnSe/ZnS QDs with the PL peak reaching 460 nm, the size of ZnSe/ZnS QDs should be sufficiently large to render the quantum confinement effect negligible. To date, the size control of ZnSe QDs has been implemented simply by varying core reaction conditions (*e.g.*, time and temperature).^{99,100} Analogous to InGaP QDs,¹⁰¹ an alternative strategy toward blue emission from ZnSe QDs is compositional alloying of

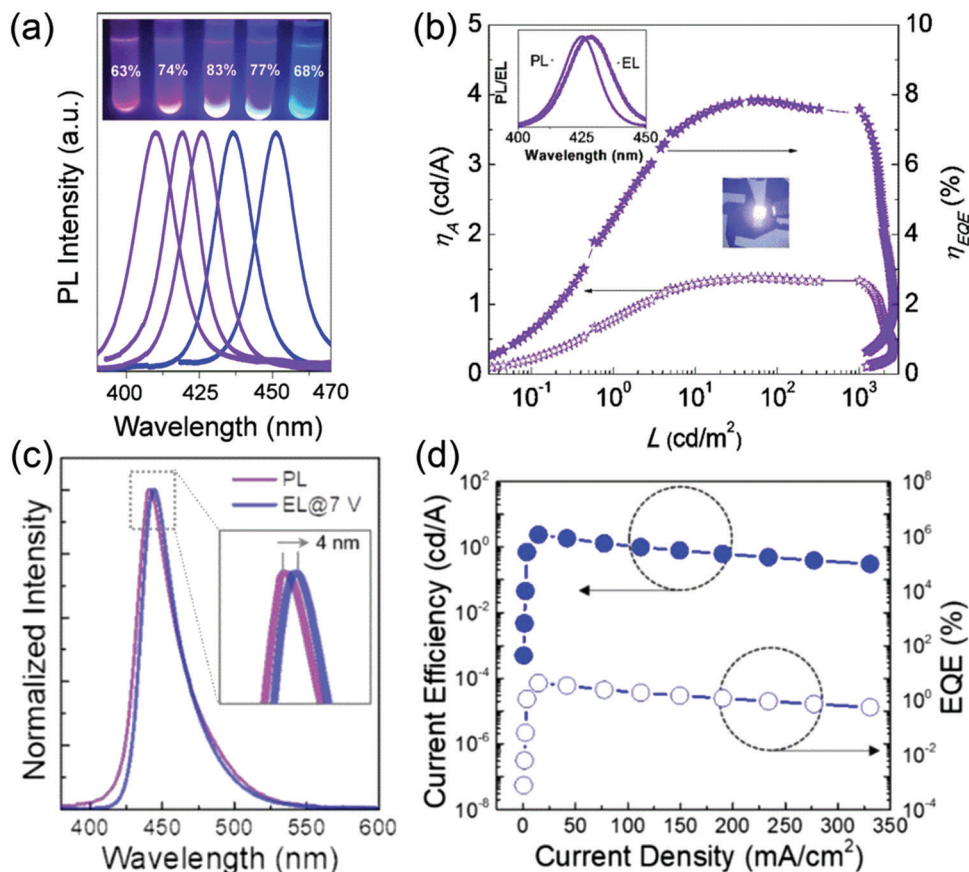


Fig. 11 (a) Normalized PL spectra of ZnSe/ZnS core/shell QDs (dispersed in hexane) and representative emission colors of ZnSe/ZnS core/shell QDs (in hexane) under the radiation of an ultraviolet lamp. (b) Current efficiency– η_{EQE} – L characteristics for the ZnSe/ZnS-based QD-LEDs. Reproduced with permission.⁹⁸ Copyright 2015, Royal Society of Chemistry. (c) Spectral comparison of the PL of a dilute ZnSeTe/ZnSe/ZnS QD dispersion with the EL collected at 7 V. (d) Current efficiency and η_{EQE} as a function of current density for QD-LEDs with an ITO/PEDOT:PSS/PVK/ZnSeTe/ZnSe/ZnS QDs/ZnMgO/Al structure. Reproduced with permission.¹⁰² Copyright 2019, American Chemical Society.

high-bandgap ZnSe with low-bandgap ZnTe in QD synthesis. Yang *et al.* developed a heat-up synthesis of alloyed ZnSeTe QDs, and further introduced inner and outer shells of ZnSe and ZnS, respectively, to form double-shelled ZnSeTe/ZnSe/ZnS QDs. As shown in Fig. 11c, the resulting blue (441 nm)-emission QDs with an optimal Te/Se ratio in the ZnSeTe core and ZnSe inner shell thickness possessed a high PL QY of 70% and a narrow FWHM of 32 nm. Finally, owing to the suppressed nonradiative processes of Förster energy transfer and Auger recombination with the presence of relatively thick double shells and ZnSe inner shell, the blue ZnSeTe QD-LEDs that utilized such QDs exhibited a peak luminance of 1195 cd m^{−2} and an η_{EQE} of 4.2% (Fig. 11d), which is the highest device performance to date among the non-Cd blue QD-LEDs. This work will pave the way for the realization of next-generation non-Cd QD-LED displays in the future.¹⁰²

4.2 InP-based QD-LEDs

Among the non-Cd QDs, InP QDs have been highlighted as the most promising alternatives to CdSe QDs because of their increasingly enhanced PL characteristics in terms of PL line width and PL QY, mainly enabled by an elaborate control of InP

core size distribution and rational multi-shell-based heterostructural engineering.^{103–105} Following years of hard work on the optimization of emitting materials, device architecture, and deep understanding of the operational mechanism, the η_{EQE} of red and green InP-based QD-LEDs has exceeded 12% and 13%, respectively;^{25,26} however, their blue-emitting counterpart has been challenging because of a relatively small bulk bandgap (1.35 eV at room temperature) of InP QDs.¹⁰⁶ In 2017, Shen *et al.* synthesized high PL QY (up to 76%) InP/ZnS small-core/thick-shell tetrahedral-shaped QDs for blue QD-LEDs for the first time. However, the maximum luminance was only 90 cd m^{−2}, and no other EL characteristics were reported.¹⁰⁷ The poor device performance may be attributed to the low PL QY of the QD assembled emitting thin film as well as the low carrier injection and transport process. Although an extra thick ZnS shell on the surface of an InP core can enhance the PL QY, ultimately suppressing the nonradiative Förster resonance energy transfer (FRET) processes between small sized QDs, the large lattice mismatch between InP and ZnS (~7.85%) will lead to the lattice collapse of InP QDs. So, in 2020, Shen *et al.* introduced a thin GaP bridging layer between the inner InP core and the outer ZnS shell through the “low temperature nucleation and high

temperature growth" method, which significantly reduced the impact of the large lattice mismatch. By further coating of the ZnS thick shell, the resultant InP/GaP/ZnS//ZnS QDs showed a large particle size of 7.0 ± 0.9 nm and a high PL QY of 81%. The corresponding QD-LEDs exhibited a record brightness and η_{EQE} of 3120 cd m^{-2} and 1.01%, respectively. Using large-scale density functional theory (DFT) calculations on thousand-atom QDs, Shen *et al.* found that thicker-shell ones favor a more balanced carrier injection and simultaneously suppress the FRET between closely packed QDs, which collectively improve the performances of blue InP-based QD-LEDs.¹⁰⁸ In addition, reports are available on blue InP-based QDs based on changing the ratio of InP:ZnS, reaction conditions, types of precursors and surface ligands, or even doping Ga ions.^{109–112} For example, Ramasamy *et al.* synthesized highly luminescent In(Zn)P/ZnSe/ZnS QDs with tunable emission from 488 to 641 nm and high color purity by synthesizing Zn carboxylate covered In(Zn)P QDs in a single step, and the FWHM and PL QY were 35 nm and 44%, respectively, for blue-sky emission (488 nm).¹¹³ In addition, blue-emitting InP/ZnS core/shell QDs with a PL peak of 475 nm and a FWHM of 39 nm were demonstrated by decreasing the reaction temperature after mixing the precursors.¹¹⁴ However, the PL peak of these QDs is mostly higher than 470 nm, and they have no application in QD-LEDs. Hence, the performance of pure blue InP-based QD-LEDs needs to be further improved.

4.3 Cu-based QD-LEDs

Cu-based QDs are another promising class of non-Cd emitting materials. By controlling composition, size, and cationic offstoichiometry, the PL emission of these QDs can be tuned from green to deep red. However, the emitting wavelengths of I–III–VI QDs shorter than 500 nm are rarely reported. Recently, a rather wider tunability has been accomplished primarily through alloying Cu–In–S QDs with a higher bandgap material of ZnS or Cu–Ga–S to produce quaternary Zn–Cu–In–S or Cu–In–Ga–S ones. For the first time, Yang *et al.* demonstrated a sky-blue Cu–Ga–S/ZnS QD-based QD-LED having a luminance of 39 cd m^{-2} and an η_{EQE} of 0.007%. The poor performance may be due to the low PL QY ($\sim 20\%$) of Cu–Ga–S/ZnS QDs.¹¹⁵ The same group also optimized Cu–Ga–S/ZnS QDs *via* Zn alloying, and the resulting Zn–Cu–Ga–S/ZnS core/shell QDs possessed tunable emissions in the azure-to-blue range and a high PL QY of 78–83% with different amounts of the Zn precursor (Fig. 12a and b). Nevertheless, no report is available on the related QD-LEDs.¹¹⁶ Based on bluish-emitting double-shelled QDs of Zn–Cu–Ga–S/ZnS/ZnS, a series of Mn doped Zn–Cu–Ga–S QDs with various Mn/Cu molar ratios were synthesized *via* surface adsorption and lattice diffusion. Regardless of the concentration of Mn-doping, all QDs possessed high PL QYs ranging from 74% to 79%. Similar to the Mn-doped Cd-based QDs, the resulting Mn-doped Zn–Cu–Ga–S QDs exhibited a distinct $\text{Mn}^{2+} {}^4\text{T}_1\text{--}{}^6\text{A}_1$ emission simultaneously with host intra-gap states involving emissions of free-to-bound and donor–acceptor pair (DAP) recombinations. However, different from Mn-doped Cd-based QD-LEDs, Mn emission was nearly absent in the EL of the Zn–Cu–Ga–S:Mn/ZnS/ZnS based QD-LED (Fig. 12d),

which may be attributed to a preferential hole injection from the HTL to the QD emitting layer and subsequent rapid trapping of holes at acceptor states. When these QDs were incorporated into QD-LEDs, the related device with Mn/Cu = 16 exhibited a high luminance of 1352 cd m^{-2} , a peak current efficiency of 2.3 cd A^{-1} , and a maximum η_{EQE} of 4.2% (Fig. 12c).¹¹⁷ Despite these parameters, Cu–In–S-based QDs with Mn-doping require detailed adjustment of the Mn concentration, increasing the complexity of the synthesis. Therefore, after two years of hard work, Yang *et al.* reported non-Cd Zn–Cu–Ga–S/ZnS QDs with a high PL QY of 80% and a PL peak at 474 nm. The related QD-LEDs demonstrated a record η_{EQE} of 7.1% and a peak current efficiency of 11.8 cd A^{-1} by all-solution-processed fabrication, corresponding to the record quantities among non-Cd QD-based blue EL devices reported to date. This work accelerates the commercialization of non-Cd QD LEDs.¹¹⁸

4.4 Carbon dot-based QD-LEDs

CDs generally comprise abundant elements and are free of heavy metals. Since the first synthesis of CDs reported by Sun *et al.* in 2006,¹¹⁹ these materials have attracted widespread research attention owing to their tunable luminescence emission, high resistance to photobleaching, and intrinsic nontoxicity. In recent years, CDs have been demonstrated to possess high PL QYs (up to 60–80%), rendering them competitors in light-emitting performance. The first blue carbon-dot based LED (CD-LED) was reported by Zhang *et al.*, who fabricated CD-LEDs by sandwiching the CD emitting layer between the organic HTL (PEDOT:PSS/poly-TPD) and the organic ETL (TPBi); this device exhibited a turn-on voltage of 5 V and a maximum brightness of 24 cd m^{-2} . However, therein, the emission would transfer from blue to white light with increasing voltage up to 9 V.¹²⁰ Hence, to obtain pure blue emission at any voltage, Yuan *et al.* synthesized multicolor bandgap fluorescent CDs by simply controlling the fusion and carbonization of citric acid (CA) and diamino-naphthalene (DAN) with a unique amino-substituted rigid carbon skeleton structure. The as-prepared CDs were nitrogen-doped and had a high PL QY of up to 75% for blue emission, a high degree of crystallinity, and high surface passivation. Consequently, the pure blue LEDs were realized by directly using these CDs as the emitting layer without the HTL, and these blue LEDs exhibited voltage-independent emission with an increased luminance of 136 cd m^{-2} . This work served the purpose of understanding and providing significant opportunities for further improvement in CD-LEDs.¹²¹ In 2018, Xu *et al.* adopted a strategy of host–guest doping to fabricate blue CD-LEDs using the oleophilic CDs (guest) doped in PVK (host) as the emitting layer because of the good co-solubility of oleophilic CDs and PVK in chlorobenzene at any doping concentration. Thus, the fabricated CD-LEDs obtained an ultra-high brightness of 569.8 cd m^{-2} with pure blue emission that predominantly originated from the emission of CDs by the host–guest energy transfer mechanism. This result demonstrated that oleophilic CDs were conducive to fabricating CD-LEDs with both high brightness and pure emission. However, the turn-on voltage of CD-LEDs was 8.5 V, which might be due to the deep HOMO of these CDs, reaching up to -6.43 eV .¹²² Moreover, the FWHM of

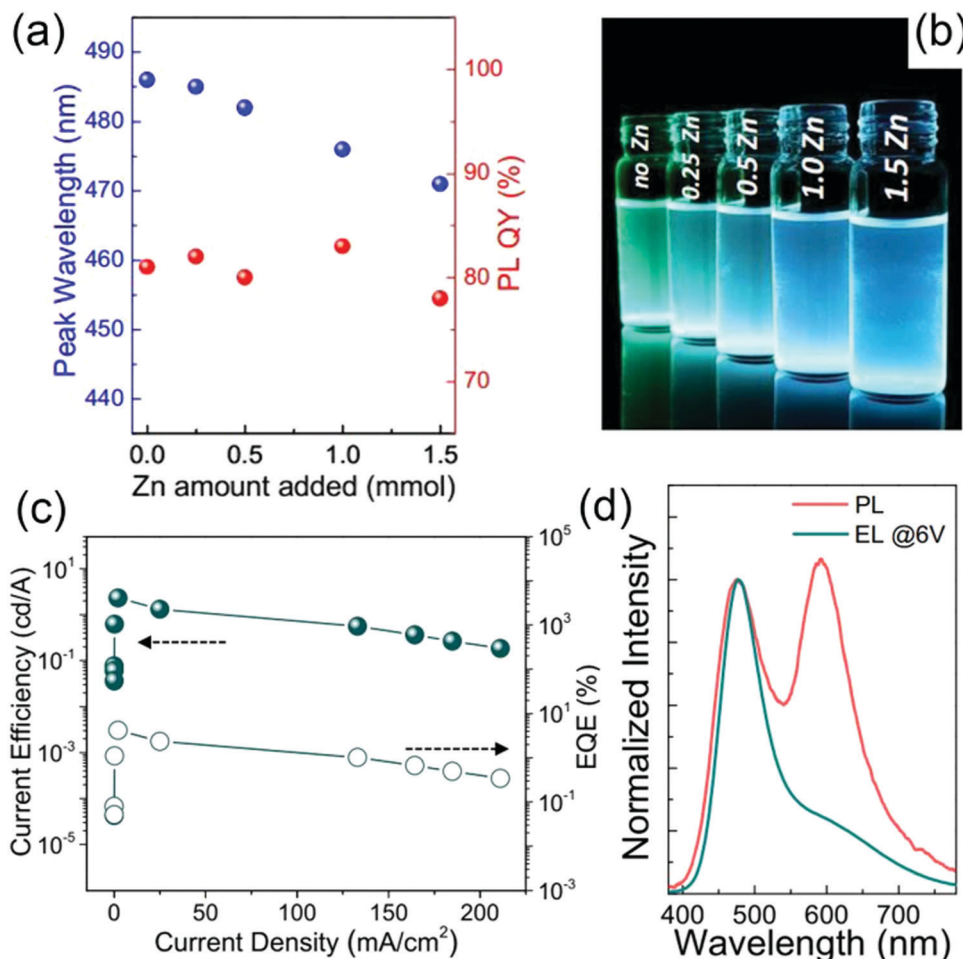


Fig. 12 (a) PL peak–PL QY of azure-to-blue-emitting Zn–Cu–Ga–S/ZnS QDs synthesized with different amounts of the Zn precursor. (b) Photograph of highly fluorescent QD dispersions under UV lamp excitation. Reproduced with permission.¹¹⁶ Copyright 2017, Royal Society of Chemistry. (c) Current efficiency– η_{EQE} –current density characteristics of the Mn/Cu = 16-doped QD-LED. Reproduced with permission.¹¹⁷ Copyright 2019, American Chemical Society.

these CDs typically exceeds 80 nm, which may originate from the electronic coupling to molecular vibrations and distortions, limiting their display applications. Therefore, Yuan *et al.* adopted the tri-molecular reaction route to design the neighboring active –OH and –H groups for six-membered ring cyclization with three-fold symmetric phloroglucinol (PG) as the reagent. Owing to the molecular purity and crystalline perfection, the as-synthesized triangular CDs exhibited high color purity with a narrow FWHM of 29–30 nm and high PL QYs of up to 54–72% (Fig. 13a). As shown in Fig. 13b and c, when incorporating these CDs into LEDs, a high-performance device having a maximum luminance of 1882 cd m⁻², a current efficiency of 1.22 cd A⁻¹, and a low turn-on voltage of 4.3 V was obtained at an EL emission peak of 476 nm. More importantly, the CD-LEDs exhibited an apparently voltage-independent emission color, indicating the high color stability of the device (Fig. 13d).¹²³ Moreover, different oxygen-containing functional groups such as carboxyl, carbonyl, and epoxy groups at the edge or basal plane sites in the CDs, functioning as transient trap states, may also contribute to spectral broadening. To eliminate the oxygen-containing functional groups

while retaining solution processability, Sargent *et al.*¹²⁴ developed a two-step method for the synthesis of high-color-purity blue CDs, in which the CDs functionalized with OH, COOH, and NH₂ were first synthesized through solvothermal treatment using citric acid and diaminonaphthalene, followed by additional surface amination using ammonia liquor and hydrazine hydrate under high temperature to further eliminate the oxygen-containing functional groups. Finally, the CDs were purified *via* silica column chromatography using a mixture of dichloromethane and methanol as the eluent to remove the byproducts and unreacted precursors. The as-prepared CDs exhibited a narrow FWHM of 35 nm and a high PL QY of 70 ± 10%, which are among the highest values for CDs reported to date. The high PL QY could be ascribed to efficient edge amination, which decreased the number of defects of CDs and thus suppressed non-radiative pathways. When incorporating these CDs blended PVK into LEDs, a maximum luminance of 5240 cd m⁻², a peak η_{EQE} of 4%, and a current efficiency of 2.6 cd A⁻¹ were obtained at an EL emission peak of 440 nm, exceeding deep-blue QD-LEDs based on Cd/Pb-comprising materials.^{50,125} However, the investigation of new materials as

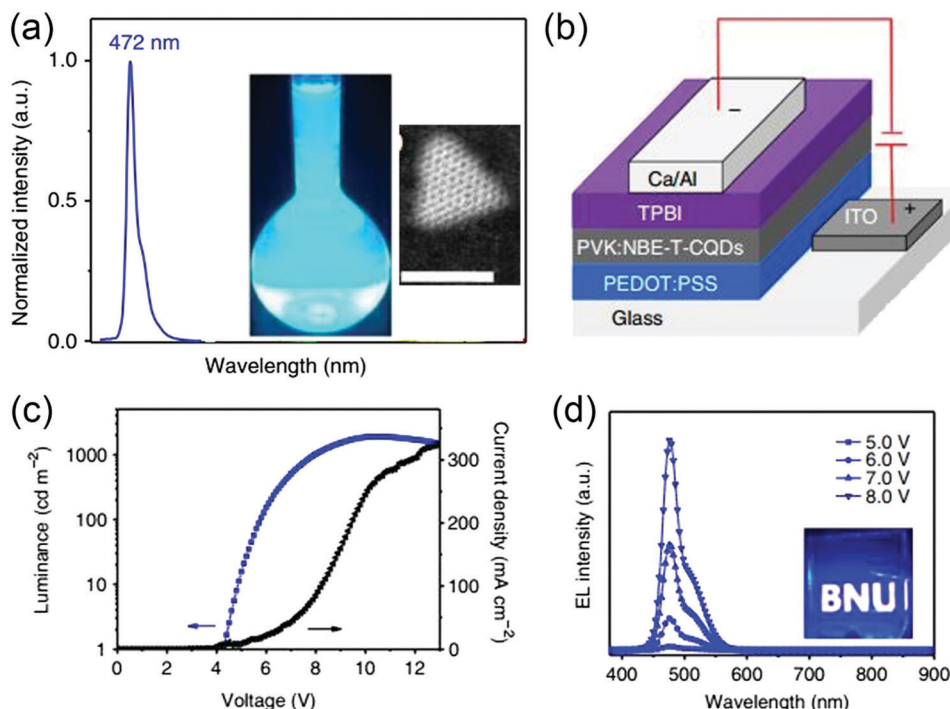


Fig. 13 (a) PL spectrum of the CDs. Insets: Photographs under UV light (left) and HAADF-STEM image of the CDs. (b) Device structure of the CD-LEDs. (c) L - J - V characteristic of the CD-LEDs. (d) EL spectra of the blue-LEDs at different bias voltages. Reproduced with permission.¹²³ Copyright 2018, Nature Publishing Group.

the emitting layer for LEDs is at an initial stage. Hence, more optimization of charge injection (holes and electrons) and better-designed devices will be put forward for high-performance CD-LEDs.

5. Conclusion and perspective

In recent years, blue QD-LEDs have experienced major development in terms of QDs and CTL materials, device physics, and novel device structures. In this review, we studied the development of blue-emitting QD-LEDs, including Cd-based QD-LEDs, PeLEDs, and heavy metal-free QD-LEDs. The Cd-based QD-LEDs could be optimized *via* nanostructure engineering and stable surface modification of the QD emitting layer, engineering of related device architecture, and a deep understanding of the aging mechanism of blue devices; the device's η_{EQE} and operating lifetime have increased from less than 0.1% to 19.8% and from almost negligible hours to 7000 h, respectively. The η_{EQE} of PeLEDs has increased from less than 0.001% to 11% through structure and component optimization of perovskite QDs and by selecting the appropriate device architecture. However, compared with their red and green counterparts, the efficiency and device lifetime of blue QD-LEDs are still lagging. In addition, the development of heavy metal-free QD-LEDs is relatively slow owing to the challenge in preparing blue emitting materials. Hence, the following aspects are critical for accelerating the commercialization of blue QD-LEDs.

(1) Performance and operating lifetime of blue Cd-based QD-LEDs

To date, the device efficiency of blue QD-LEDs has remained lower than that of the red and green devices (red: $\eta_{\text{EQE}} = 30.9\%$,

green: $\eta_{\text{EQE}} = 25.04\%$).^{126,127} Additionally, the longest operation lifetimes of Cd-based blue QD-LEDs are only on the order of 7000 h at an initial brightness of 100 cd m^{-2} , which is a little inferior to the minimum requirement for display applications (10 000 h). These poor device performances may be attributed to the relatively low PL QYs of QD films, the imbalanced charge injection/transport resulting from the large energy level offset between the QD layers and conventional organic HTLs with lower hole mobility, and the vague understanding of quenching mechanisms, such as energy transfer in blue QD films, interfacial charge transfer between blue QDs and CTLs, and electrical-field-induced quenching in blue QD-LEDs.^{30,31,128} Therefore, to develop efficient blue QDs and related QD-LEDs, intensive research is required on crystal engineering for blue QDs to have the appropriate energy-level structure, material chemistry for high-performance HTLs with simultaneous high work function and hole mobility, together with an in-depth understanding of the exciton quenching mechanism of blue QDs.

(2) Stability of blue PeLEDs

The stability and lifetime of perovskite LEDs remain a challenge, hindering their practical application. Although efforts have been directed toward improving the stability of perovskite devices by mixing inorganic Cs cations and organic cations at the A site and constructing 2D or quasi-2D perovskites as well as modifying the perovskite surface by appropriate ligands, exciton quenching still exists at the interface between CTLs and QDs. Currently, the best operating lifetime of blue perovskite devices is 10 min under a constant applied voltage of 4.4 V, which is far from the requirement of display and lighting applications. This poor

performance may be associated with ion migration and poor thermostability in the perovskite. Therefore, further research is required to address issues such as improving the reproducibility of perovskite devices, long-term stability, and a deep fundamental understanding of lead halide perovskite materials and their device architectures.

(3) Development of heavy-metal-free blue QD-LEDs

Despite the available high performance, Cd- and perovskite-based QDs contain toxic elements Cd and Pb whose amount should be tightly controlled in consumer electronics. Hence, synthesizing heavy-metal-free QDs is highly desirable. In this case, several non-Cd materials, such as ZnSe, InP, Cu-based QDs, and carbon dots, have been investigated in EL devices.^{98,107,118,123} Nevertheless, the EL peaks of these devices are mostly above 480 nm, and the device performance shows a huge margin with conventional QD-LEDs using Cd-based QDs. Therefore, researchers must urgently optimize synthetic methods to synthesize nanocrystals with few defects and uniform sizes as well as device architectures to balance charge injection and transport in devices.

(4) Applications of blue QD-LEDs in lightings and displays

Currently, there is no precedent for mass production of any solution-processed LEDs. The application of QDs with large-area uniformity and high PL QYs in lightings and displays may require flexible plastic substrates and low manufacturing cost of the QD-LEDs. However, QD-LEDs based on flexible plastic substrates show a huge margin in device performance based on the requirements of lightings and displays, and the flexible plastic substrates introduce additional problems, such as encapsulation. Furthermore, the large-scale synthesis of colloidal QDs has rarely been reported, which increases the manufacturing costs of the QD emitting layer. Extensive efforts should be devoted to finding a highly flexible polymeric film (*e.g.*, polyethylene naphthalate (PEN), polyethylene terephthalate (PET), or polyimide (PI)) and a highly transparent metal electrode (*e.g.*, Ag nanowire or graphene) to realize flexible and transparent devices.^{129–133} Moreover, new synthesis methods should be developed to synthesize blue QD emitting materials on a large scale.

(5) Perspective

Finally, because they are similar to OLEDs, QD-LEDs possess a high freedom of device design, a fascinating aspect for next-generation displays; they can even rival OLEDs and micro-LEDs, and we look forward to seeing the successful commercialization of QD-LED displays in the near future.

Conflicts of interest

The authors declare no competing financial interest.

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