



Precisely Controlled yet Dynamically Exchanged Micelles via the Self-Assembly of Amphiphilic Acrylate Random Copolymers in Water

Journal:	Polymer Chemistry
Manuscript ID	PY-ART-11-2024-001272.R1
Article Type:	Paper
Date Submitted by the Author:	19-Dec-2024
Complete List of Authors:	Kono, Hiroyuki; Kyoto University Faculty of Engineering Graduate School of Engineering Department of Polymer Chemistry, Ouchi, Makoto; Kyoto University Faculty of Engineering Graduate School of Engineering Department of Polymer Chemistry, Terashima, Takaya; Graduate School of Engineering, Kyoto University, Department of Polymer Chemistry

SCHOLARONE™ Manuscripts

ARTICLE

Received 00th January 20xx, Accepted 00th January 20xx

DOI: 10.1039/x0xx000000x

Precisely Controlled yet Dynamically Exchanged Micelles via the Self-Assembly of Amphiphilic Acrylate Random Copolymers in Water

Hiroyuki Kono,^a Makoto Ouchi,^a and Takaya Terashima*^a

Herein, we investigated the self-assembly of amphiphilic acrylate random copolymers bearing hydrophilic poly(ethylene glycol) chains and hydrophobic dodecyl groups into micelles in water. The random copolymers formed precise yet dynamic micelles in water, dependent on degree of polymerization (DP) and composition. The copolymers shorter than a threshold DP_{th} exclusively formed multichain micelles and the copolymers longer than the DP_{th} self-folded into unimer micelles. The molecular weight and size of the multichain micelles were determined by the composition, and the aggregation number was controllable by the DP. Critical micelle concentration of the random copolymers was estimated to be approximately 1×10^{-3} mg/mL, and almost independent of the DP, aggregation number, monomer sequence, and backbone structures. More uniquely, owing to the flexible backbones, the acrylate random copolymer micelles induced the exchange of polymer chains even at a low temperature such as 10 °C (activation energy: $E_a = ^{\sim}40$ kJ/mol) although corresponding methacrylate counterparts with relatively rigid backbones required at least 25 °C for polymer chain exchange.

Introduction

Self-assembly of amphiphilic polymers and molecules in water is a key strategy to produce nanostructured objects such as micelles, vesicles, and single-chain polymer nanoparticles. 1-¹⁶ These self-assemblies are useful as functional materials including materials, 17-19 stimuli-responsive molecular encapsulation/release materials, 20,21 and drug-delivery vehicles.^{22,23} In general, polymer or surfactant micelles and related aggregates are formed in water via hydrophobic effects or physical interaction above critical micelle or aggregation concentration. Those self-assemblies often induce the exchange of the polymer chains or molecules each other; the dynamic behaviour depends on the molecular structures, solution temperature, solute concentration, and additives in solutions. Controlling not only the size and three-dimensional architectures but also dynamic properties is crucial in designing self-assembled materials with desired properties and functions.

To date, various amphiphilic polymers with distinct structures such as block, 1-3,5-8,24-27 random/statictical, 28-43 alternating 44-48 copolymers, have been designed for targeted self-assemblies. Among them, random copolymers have attracted attention as scaffolds for small micelles or SCNPs whose size is about 10 nm and close to proteins. We have developed self-assembly

systems of amphiphilic random copolymers bearing hydrophilic poly(ethylene glycol) (PEG) and hydrophobic alkyl groups [e.g., PEG methyl ether methacrylate (PEGMA)/dodecyl methacrylate (DMA) random copolymers (Figure 1)]. These copolymers form micelles via chain-folding by the association of hydrophobic side chains in water and show unique self-assembly behaviour in water, depending on the degree of polymerization (DP): the copolymers shorter than a threshold DPth exclusively form multichain micelles via intermolecular self-assembly and those longer than the DPth form unimer micelles via self-folding. The size of the multichain micelles is determined by the copolymer composition and side chain structures, irrespective of DP. Additionally, the random copolymer micelles show dynamic chain-exchange behaviour, depending on the side chains and temperature. 41-43

Self-assembly of amphiphilic (co)polymers is also dependent on the main chain structures and monomer sequence. For instance, the alternating copolymer of PEG methyl ether acrylate (PEGA) and dodecyl vinyl ether (DVE)48 forms a multichain micelle with molecular weight lower than corresponding methacrylate³⁵ or acrylate³⁶ random copolymers with the same composition of PEG and dodecyl side chains. This is probably because the alternating copolymer has a flexible acrylate/vinyl ether backbone and does not contain consecutive dodecyl units. Acrylamide random copolymers consisting of more hydrophilic backbones also formed micelles with molecular weights lower than acrylate counterparts.³⁸ These results suggest that flexibility and polarity of the polymer backbones affect dynamic properties of polymer micelles. Owing to the flexible backbones, amphiphilic acrylate random copolymers^{36,37} are expected to induce precise yet dynamic self-

^a Department of Polymer Chemistry, Graduate School of Engineering, Kyoto University, Katsura, Nishikyo-ku, Kyoto 615-8510, Japan. E-mail: terashima.takaya.2e@kyoto-u.ac.jp

[†] Footnotes relating to the title and/or authors should appear here. Supplementary Information available: [Experimental details for the synthesis and characterization of polymers, SEC, NMR, DLS, and fluorescence spectroscopy]. See DOI: 10.1039/x0xx00000x

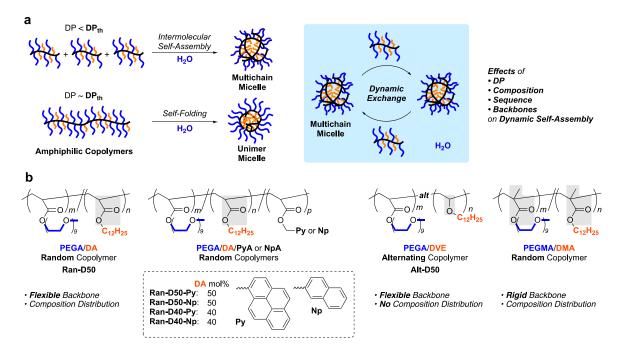


Figure 1. (a) Precision self-assembly of amphiphilic copolymers into multichain or unimer micelles in water, depending on the degree of polymerization (DP), and dynamic exchange of polymer chains between micelles. (b) Design of acrylate random, acrylate/vinyl ether alternating, and methacrylate random copolymers with hydrophilic poly(ethylene) glycol (PEG) chains and hydrophobic dodecyl groups.

assembly into size-controlled micelles that may promote chain exchange more than methacrylate copolymer micelles. 41-43

Herein, we investigated the self-assembly of amphiphilic acrylate random copolymers bearing PEG chains and dodecyl groups into micelles in water, focusing on the effects of the DP, composition, and sequence distribution on the size and aggregation number of the micelles, critical micelle concentration (CMC), and chain exchange properties (Figure 1). The random copolymers were obtained from free radical copolymerization of hydrophilic PEGA, hydrophobic dodecyl acrylate (DA) in the presence or absence of the small amounts of pyrene (Py) or naphthalene (Np)-bearing acrylates (PyA or NpA). The copolymers with broad dispersity (£) were fractionated into samples with different molecular weight and narrow \mathcal{D} by preparative size-exclusion chromatography (SEC). The aqueous solutions of the fractionated copolymers were analyzed by SEC equipped with multiangle laser light scattering (MALLS) to determine the absolute weight-average molecular weight and aggregation number of the micelles. The chain exchange between their micelles was evaluated by fluorescence measurements of the mixtures of the Py or Np-labeled copolymers.

The acrylate random copolymers, as well as methacrylate counterparts, 35,39,40 induced self-assembly controlled by the DP and composition. The copolymers with DP smaller than a threshold DPth formed multichain micelles whose size was constant and independent of DP. In contrast, the copolymers with DP larger than DPth mainly formed unimer micelles. The size of multichain micelles increased with increasing the content of dodecyl groups. CMC of the random copolymers in water was estimated to be approximately 1.0 \times 10-3 mg/mL, independent of DP (i.e., aggregation number) and monomer

sequence distribution (random or alternating). Uniquely, the acrylate random copolymers induced chain exchange even at a low temperature more efficiently than methacrylate counterparts.

Results and discussion

Synthesis of Amphiphilic Random Copolymers.

Amphiphilic random copolyacrylates bearing hydrophilic PEG chains, hydrophobic dodecyl groups, and the small amount (~1 mol%) of pyrene (Py) or naphthalene (Np) units were synthesized by free radical copolymerization of PEGA, DA, PyA or NpA with 2,2′-azobis(isobutyronitrile) in toluene at 60 °C. The DA content was set to 40 or 50 mol% to examine the effects of composition, main chain structures, and monomer sequence (random vs. alternating) on self-assembly behaviour. The Py or Np fluorophores were used to evaluate the exchange of the polymer chains between their micelles in water by fluorescence resonance energy transfer (FRET). The random copolymers are coded as Ran-D40-Py or Np and Ran-D50-Py or Np, dependent on the DA content and the Py or Np labels. A non-labeled PEGA/DA random copolymer with 50 mol% DA (Ran-D50) was also prepared to investigate CMC in water.

In all the copolymerization, both PEGA and DA were simultaneously consumed at the same rate, regardless of the feed ratio of their monomers. This indicates that the two monomers are randomly distributed in resulting copolymer chains (Figure S1). The random copolymers were analyzed by SEC in N'N'-dimethylformamide (DMF) containing 10 mM LiBr. The copolymers had number-average molecular weight (M_n) of 23,400 - 34,000 g/mol and dispersity ($D = M_w/M_n$: molecular weight distribution) of 2.05–2.43 by poly(methyl methacrylate)

(PMMA) standard calibration (Figures 2a and S1) or M_n of 10,500 - 15,600 g/mol and $\mathcal D$ of 2.36–2.86 by poly(ethylene oxide) (PEO) standard calibration (Figures S2 and S3). To examine the effects of the monomer sequence distribution and backbone structures on CMC, a PEGA/DVE alternating copolymer (**Alt-D50**, M_n = 33,500 g/mol, $\mathcal D$ = 1.72 by PMMA calibration or M_n = 15,600 g/mol, $\mathcal D$ = 1.89 by PEO calibration) was also prepared by free radical copolymerization of PEGA in the presence of an excess of DVE according to the literature. ⁴⁶ The alternating sequence of PEGA and DVE was confirmed by 13 C NMR spectroscopy (Figure S4).

All the copolymers with broad dispersity were fractionated using a preparative SEC into six samples (A–F) with different molecular weights (Figures 2, S2, and S3). The chemical structures and compositions of the fractionated copolymers were analyzed by 1 H NMR spectroscopy (Figures S5–S10, Tables S1–S4). The absolute weight-average molecular weight ($M_{\rm w,DMF}$) and $\mathcal{D}_{\rm DMF}$ of the fractionated copolymers were determined by SEC equipped with a MALLS detector in DMF (10 mM LiBr) as an eluent, where the copolymers are unimolecularly dissolved in

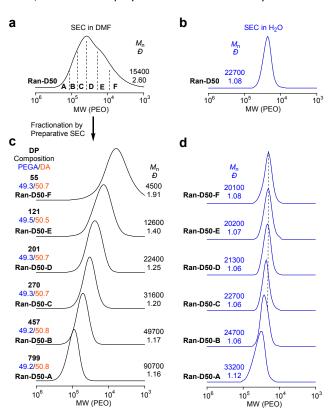


Figure 2. SEC curves of **Ran-D50** in (a) DMF (10 mM LiBr) and (b) H_2O (100 mM NaCl) with PEO standard calibration. **Ran-D50** was fractionated into six samples with different molecular weights (**A-F**) by preparative SEC. (c, d) SEC curves of fractionated **Ran-D50-A – F** with different DPs in (c) DMF (10 mM LiBr) and (d) H_2O (100 mM NaCl). The in-set M_0 and D values are determined with PEO standard calibration.

DMF. The degree of polymerization (DP) of the fractionated copolymers was calculated from the $M_{\rm w,DMF}$ and $\vartheta_{\rm DMF}$ (by SEC-MALLS), the composition (by $^1{\rm H}$ NMR), and the formula weight of monomers (Tables S1–S4). Typically, Ran-D50-F – A ($M_{\rm n}$ =

4,500 - 90,700 g/mol, $D = 1.16 - 1.91 \text{ by SEC with PEO calibration, Figure 2c) had DP of <math>55 - 799$.

Self-Assembly of Amphiphilic Random Copolymers into Micelles in Water.

Self-assembly of amphiphilic random copolymers (Ran-D50, Ran-D50-Py or Np, Ran-D40-Py or Np) into micelles in water was evaluated using SEC-MALLS in H₂O containing 100 mM NaCl as an eluent. The aqueous solutions of the copolymers were prepared as follows: In vials, the copolymers were dissolved in water at 25 °C (1 mg/mL), giving transparent solutions. The solutions were sonicated at 25 °C for 3 min (sonicator: Branson, Bransonic 1510) filtered and poly(tetrafluoroethylene) membrane filter (0.45 μm, Merck Millipore) before analysis. The aqueous solutions were injected into the SEC system to determine M_n (apparent size), absolute weight-average molecular weight ($M_{w,H2O}$), and $\mathcal D$ of the polymer micelles in water (Tables S1-S4). The apparent size of the micelles was evaluated with M_n determined by SEC in water with PEO calibration. $M_{w,H2O}$ of the micelles was determined by SEC-MALLS in water to estimate the aggregation number (N_{agg}) of the copolymers [$N_{agg} = M_{w,H2O}$ (MALLS)]/ $M_{w,DMF}$ (MALLS)]. θ of

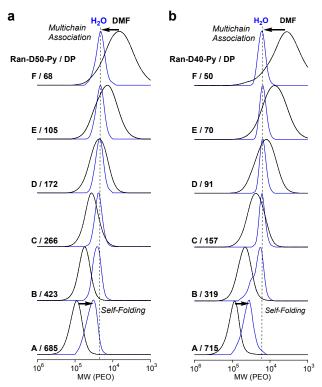


Figure 3. SEC curves of (a) Ran-D50-Py-A – F and (b) Ran-D40-Py-A – F with different molecular weights in DMF (black lines) and H_2O (blue lines).

the micelles was determined by PEO calibration or MALLS.

Effects of DP on Apparent Micelle Size. The aqueous solution of Ran-D50 with broad dispersity ($\mathcal{D}=2.60$ in DMF, Figure 2a) was first analyzed by SEC in water containing 100 mM NaCl. Ran-D50 showed a unimodal SEC curve with narrow dispersity in water ($\mathcal{D}=1.08$, Figure 2b). This implies that the acrylate-based Ran-D50, as well as methacrylate-based PEG/dodecyl

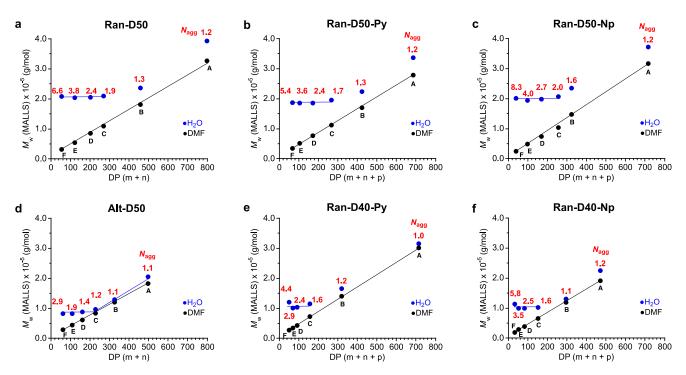


Figure 4. Absolute weight-average molecular weight (M_{w} , determined by SEC-MALLS) of (a) Ran-D50-A – F, (b) Ran-D50-Py-A – F, (c) Ran-D50-Np-A – F, (d) Alt-D50-A – F, (e) Ran-D40-Py-A – F, and (f) Ran-D40-Np-A – F in water (100 mM NaCl, blue) or DMF (10 mM LiBr, black) as a function of DP. Aggregation number: $N_{agg} = M_{w, H2O}/M_{w, DMF}$.

random copolymers³⁵ and PEGA/DVE alternating copolymers,⁴⁸ form micelles whose apparent size is primarily determined by the composition. Then, fractionated Ran-D50-A - F with different DPs were analyzed by SEC in water (Figure 2d). Ran-**D50-F – C** with 55 – 270 DP showed unimodal SEC curves with almost identical M_n (20,100 – 22,700 g/mol) and narrow dispersity (1.06 - 1.08) by PEO calibration, whereas Ran-D50-B and A with DP more than 457 showed the shift of their SEC curves to high molecular weight with increasing DP. This result suggests that Ran-D50 with DP smaller than 270 form multichain micelles whose size is independent of DP and Ran-D50 with DP larger than at least about 450 mostly form unimer micelles whose size increased with DP. Fractionated Ran-D50-Py or Np with DP smaller than approximately 270 also showed SEC curves with almost identical M_n (Figures 3a, S13, and S14). This supports that the small amount (~1 mol%) of Py or Np labels hardly affects the self-assembly behaviour of the copolymers and the apparent size of the resulting micelles. In contrast, the copolymer composition affected the self-assembly and apparent size of micelles. Typically, Ran-D40-Py-F - C with DP smaller than 160 showed SEC curves with almost identical $M_{\rm n}$ (Figure 3b), whereas the $M_{\rm n}$ was smaller than that for constant size micelles of Ran-D50(-Py or Np). In addition, the threshold DP of the constant size micelles for Ran-D40-Py was smaller than that for Ran-D50-Py.

Absolute Molecular Weight and Aggregation Number. $M_{\rm w,H2O}$ of the polymer micelles was determined by SEC-MALLS in water. $M_{\rm w,H2O}$ for Ran-D50-F – C with 55 – 270 DP was nearly constant around 200,000 g/mol and their $N_{\rm agg}$ values decreased from 6.6 to 1.9 with increasing DP (Figure 4a). In contrast, $M_{\rm w,H2O}$ for Ran-

D50-B and A with DP more than 457 increased with increasing DP and their N_{agg} values were close to 1. As confirmed by dynamic light scattering in water, hydrodynamic radius (R_h) of Ran-D50-F - C micelles was also almost constant in the range of 5.7 - 6.1 nm, independent of DP of the copolymers, whereas R_h for Ran-D50-B and A micelles increased from 6.2 nm to 7.7 nm with increasing DP (Figure S11 and Table S1). These results demonstrate that, as implied by $M_{\rm n}$ (apparent size by PEO calibration), Ran-D50 copolymers have a threshold DP (DPth) of approximately 400 between multichain micelles and unimer micelles: (1) the copolymers with DP smaller than the DP_{th} exclusively induce the intermolecular association of the polymer chains to form multichain micelles ($N_{agg} > 2$) with constant $M_{\rm w,H2O}$ and $R_{\rm h}$, where $N_{\rm agg}$ of the micelles decreases with increasing DP. (2) The copolymers with DP larger than the DP_{th} mostly induce self-folding into unimer micelles whose $M_{\rm w,H2O}$ and $R_{\rm h}$ increase with increasing DP. Py or Np-labeled Ran-D50-F-C with DP smaller than DP_{th} also formed multichain micelles with $M_{\rm w.H2O}$ (~200,000 g/mol) that is close to nonlabelled Ran-D50-F - C (Figure 4b,c).

In contrast, Ran-D40-Py (or Np)-F – C with DP smaller than about 160 formed multichain micelles with constant $M_{\rm w,H2O}$ of approximately 100,000 g/mol and Ran-D40-Py (or Np)-B and A with DP larger than about 300 formed unimer micelles whose $M_{\rm w,H2O}$ increased with increasing DP (Figure 4e,f). The constant $M_{\rm w,H2O}$ and DP_{th} of their Ran-D40 multichain micelles were smaller than those of Ran-D50 counterparts, indicating that more hydrophilic polymers form smaller micelles. This trend is consistent with the self-assembly of methacrylate-based amphiphilic random copolymers into micelles. 35,39 Furthermore,

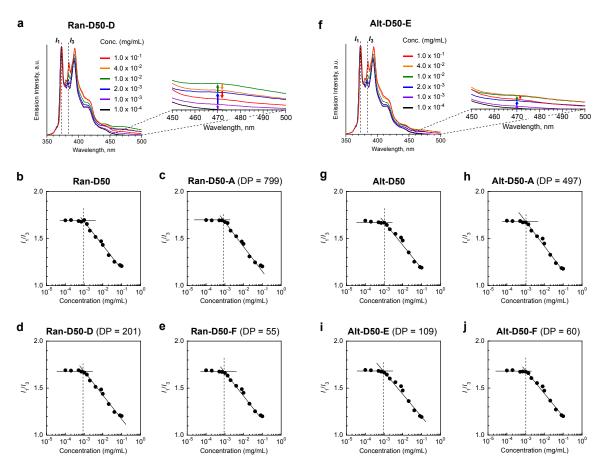


Figure 5. CMC of amphiphilic random or alternating copolymers in water. (a, f) Emission spectra of pyrene (5×10^{-7} M: normalized by I_1) containing (a) Ran-D50-D and (f) Alt-D50-E in water. (b–e, g–j) I_1/I_3 plots as a function of the polymer concentration ($1.0 \times 10^{-4} - 1.0 \times 10^{-1}$ mg/mL) of (b) Ran-D50-A, (c) Ran-D50-A, (d) Ran-D50-D, (e) Ran-D50-F, (g) Alt-D50-A, (i) Alt-D50-B, and (j) Alt-D50-F.

the constant $M_{\rm w,H2O}$ of **Alt-D50** multichain micelles (~90,000 g/mol, Figure 4d) was smaller than that of **Ran-D50** counterparts (~200,000 g/mol). This is probably because the alternating copolymers have flexible vinyl ether backbones and have no consecutive sequence of dodecyl side chains to afford the efficient folding of polymer chains into more compact micelles.⁴⁸

Critical Micelle Concentration of Amphiphilic Random or Alternating Copolymers

We examined the effects of monomer sequence and DP (related to $N_{\rm agg}$) on CMC of amphiphilic random or alternating copolymers. **Ran-D50** and **Alt-D50**, both of which have no fluorescent labels, were used. CMC of the copolymers in water was determined by fluorescence spectroscopy with pyrene as follows: 41,49,50 the aqueous mixtures of the copolymers with different concentrations $(1.0\times10^{-4}-1.0\times10^{-1}\text{ mg/mL})$ and a small amount of pyrene $(5\times10^{-7}\text{ M})$ were prepared, and the fluorescence spectra of the mixtures were measured at 25 °C (Figures 5 and S15). The emission intensity ratio of I_1 (at 373 nm) and I_3 (at 384 nm) $[I_1/I_3]$ was plotted as a function of the polymer concentration. CMC was defined as a concentration at which

 I_1/I_3 starts to decrease with increasing the polymer concentration (Figure 5).

As typically shown in Figure 5a, the I_3 intensity (normalized by I_1) for the agueous solutions of **Ran-D50-D** with DP 201 was almost constant up to approximately 1 × 10⁻³ mg/mL and increased with increasing the polymer concentration above 1 × 10-3 mg/mL. From the intersection concentration of the constant or decreasing I_1/I_3 values, CMC was determined to be approximately 1 \times 10 $^{\text{-3}}$ mg/mL (Figure 5d). Similarly, CMC of Ran-D50 with broad dispersity, Ran-D50-A with DP 799, and Ran-D50-F with DP 55 was approximately 1 × 10⁻³ mg/mL (Figure 5b,c,e). Though Ran-D50-A, D, and F form micelles with N_{agg} of 1.2, 2.4, and 6.6 in water, respectively, CMC was independent of their N_{agg} values. This is consistent with the fact that a **Ran-D50** micelle containing broad N_{agg} distribution due to broad dispersity (θ = 2.60) also have a close CMC. CMC of amphiphilic alternating copolymers [Alt-D50 with broad dispersity, Alt-D50-A with DP 497, Alt-D50-E with DP 109, and **Alt-D50-F** with DP 60] as also estimated to be 1×10^{-3} mg/mL, independent of N_{agg} (Figure 5f–j). These CMC values were close to those of poly(styrene-ethylene oxide) block copolymer micelles.50

In general, amphiphilic copolymers bearing hydrophilic and hydrophobic segments in water are placed at the air/water interface and dispersed as unimer chains or self-assemblies of multiple polymer chains in water. The three modes are dynamically exchanged in equilibrium state, depending on the polymer concentration (Figure 6a). At very low concentration, amphiphilic polymer chains are primarily located at the air/water interface, where the hydrophobic alkyl groups with low surface free energy are directed to the air. As the concentration increases, the amount of polymer chains at the interface increases. Once the interface is saturated, the polymer chains are not only dispersed as unimer chains in water but also form self-assembles such as micelles via the association of hydrophobic groups above a concentration called CMC.

CMC of both the acrylate-random and acrylate/vinyl ether alternating copolymers ($^{\sim}1.0 \times 10^{-3}$ mg/mL) was close to that of a methacrylate-random copolymer bearing PEG chains and dodecyl groups.41 This result importantly demonstrates that CMC of their amphiphilic copolymers is mainly determined by the structure and composition of the side chains, and independent of the backbone structures and monomer sequence. This is because the mass of copolymer chains potentially filling at the air/water interface and averaged hydrophobic/hydrophilic balance of the copolymer chains are independent of the DP and monomer sequence as far as the composition (the molar ratio of PEG and dodecyl groups) of the copolymers is identical. Another interesting finding is that unimer micelles of Ran-D50-A or Alt-D50-A also have CMC (~1.0 \times 10⁻³ mg/mL) as far as the fluorescence measurements using pyrene. This implies that the amphiphilic copolymers are primarily placed at the air/water interface below CMC (< $^{\sim}1.0 \times$ 10⁻³ mg/mL) and immediately form unimer micelles in water above the concentration.

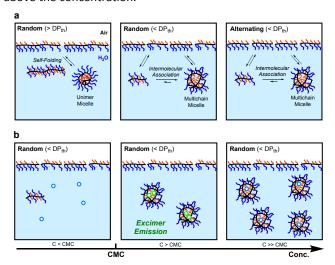


Figure 6. (a) Equilibrium of amphiphilic random or alternating copolymers in water among three states: polymer chains placed at the air/water interface and unimer chains/micelles or multichain micelles dispersed in water. (b) Effects of concentration on the encapsulation of pyrene molecules into micelles in water during CMC measurements.

The aqueous solutions of Ran-D50 and Alt-D50 further showed fluorescence intensity stemming from the excimer

emission of pyrene at around 470 nm (I₄₇₀) above their CMC ($\sim 1.0 \times 10^{-3}$ mg/mL). However, the emission intensity changed, in response to the polymer concentration (Figure 5a,f). For example, I_{470} of Ran-D50-D increased with increasing the concentration up to 1.0×10^{-2} mg/mL and again decreased with increasing the concentration. This suggests that the number of pyrene molecules enclosed within each micelle changes as the number of micelles increases (Figure 6b). When the polymer concentration is below CMC, pyrene is molecularly dispersed in water phase. This is also confirmed by the fact that the I_1/I_3 ratio of the aqueous solutions of pyrene containing the copolymers (blow CMC) is identical that of the aqueous solution of pyrene alone (5 \times 10⁻⁷ M, Figure S15). At the concentration slightly above the CMC, a few micelles formed in water enclose multiple pyrene molecules within the hydrophobic cores, leading to excimer emission. At concentration much higher than the CMC, the pyrene molecules were fully dispersed within the multiple micelle cores, decreasing the excimer emission. These results indicate that amphiphilic random or alternating copolymer micelles can encapsulate hydrophobic molecules within their cores and potentially are applicable as nanocapsules.

Dynamic Exchange of Polymer Chains between Micelles.

Dynamic properties of Ran-D50 or Ran-D40 micelles in water were investigated, focused on the effects of copolymer composition (dodecyl groups: 50 or 40 mol%), DP (related to N_{agg}), and temperature on the exchange of the acrylatecopolymer chains. The chain exchange was analyzed by FRET in the mixtures of a Py-labeled micelle and a Np-labeled micelle.41,42 The aqueous solution of a Py-labeled micelle was mixed with that of a Np-labeled micelle ([polymer] = 1.0 mg/mL) (Figure 7a). Then, fluorescence measurement of the resulting mixture was immediately started at various temperatures (10-35 °C), where the excitation wavelength for the Np units was set at 290 nm. The fluorescence intensity was monitored at 336 nm for the Np units (I_{Np}) and at 396 nm for the Py units (I_{Py}) . Förster radius of Np and Py is 2.86 nm.³⁰ Since the labeled Np or Py units are located within the hydrophobic cores of small micelles (R_h = ~ 6 nm), the coexistence of Py-labeled copolymers and Nplabeled copolymers within single micelles can be evaluated by

For example, in the mixture of a Ran-D50-Py-D (DP = 172, $N_{\rm agg}$ = 2) micelle and a Ran-D50-Np-D (DP = 170, $N_{\rm agg}$ = 2) micelle at 35 °C, $I_{\rm Py}$ increased and $I_{\rm Np}$ slightly decreased with increasing measurement time (Figure 7b). This indicates that FRET from the Np units to the Py units occurs with time progress: namely, the Np-labeled polymer chains are gradually mixed with the Py-labeled counterparts to form Np- and Py-mixed micelles in water. Additionally, the mixture of Ran-D40-Py-D (DP = 91, $N_{\rm agg}$ = 2) micelle and Ran-D40-Np-D (DP = 82, $N_{\rm agg}$ = 2) or that of Ran-D50-Py-F (DP = 68, $N_{\rm agg}$ = 5) and Ran-D50-Np-F (DP = 38, $N_{\rm agg}$ = 8) also showed similar increase of $I_{\rm Py}$ and decrease of $I_{\rm Np}$ with time progress (Figures S16-S18). The chain exchange kinetics were thus evaluated by $I_{\rm Py}$ values of the mixtures normalized over time $\{I_{\rm Py}(t)\}$ using eq 1²⁴:

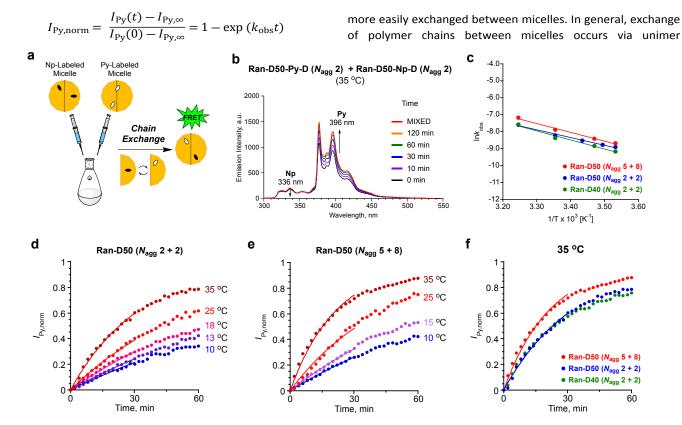


Figure 7. (a) Evaluation of polymer chain exchange between Py-labeled micelles and Np-labeled micelles by FRET. (b) Fluorescence spectra obtained from the aqueous mixture of a Ran-D50-Py-D micelle and a Ran-D50-Np-D micelle by the excitation at 290 nm at 35 °C: [polymer] = 1.0 mg/mL. The fluorescence intensity was monitored at 396 nm (I_{Py}) and 336 nm (I_{Np}) . (c) Arrhenius plots of k_{obs} for the chain exchange of the mixtures of (blue) Ran-D50-Py-D $(N_{agg} = 2)$ and Ran-D50-Np-D $(N_{agg} = 2)$, (green) Ran-D40-Py-D $(N_{agg} = 2)$ and Ran-D50-Np-F $(N_{agg} = 2)$, and (red) Ran-D50-Py-F $(N_{agg} = 5)$ and Ran-D50-Np-F $(N_{agg} = 8)$. (d, e) Effect of temperature on the chain exchange: Normalized I_{Py} $(I_{Py,norm})$ and fitting curves for the chain exchange of the binary mixtures of (d) Ran-D50-Py-D micelle $(N_{agg} = 2)$ and Ran-D50-Np-D micelle $(N_{agg} = 2)$ or (e) Ran-D50-Py-F micelle $(N_{agg} = 5)$ and Ran-D50-Np-F micelle $(N_{agg} = 8)$ in water at 10–35 °C. (f) Effect of N_{agg} and composition on the chain exchange at 35 °C: the mixtures of (blue) Ran-D50-Py-D $(N_{agg} = 2)$ and Ran-D50-Np-D $(N_{agg} = 2)$, (green) Ran-D40-Py-D $(N_{agg} = 2)$ and Ran-D50-Np-F $(N_{agg} = 5)$ and Ran-D50-Np-F $(N_{agg} = 5)$ and Ran-D50-Np-F $(N_{agg} = 8)$.

where $I_{\rm Py}(0)$ is the emission intensity from the Py units just after mixing (t=0), and $I_{\rm Py,\infty}$ is the intensity in the fully mixed state achieved after heating the mixture at 50 °C for 24 h, followed by setting to the measurement temperature. $I_{\rm Py,norm}$ was plotted against time at various temperatures.

In all the cases, $I_{Py,norm}$ increased with time, and the increase of $I_{Py,norm}$ turned fast by increasing temperature (Figure 7). This indicates that the exchange of the polymer chains is promoted upon heating the solutions. The initial $I_{\rm Py,norm}$ values were fitted by the equation (1) to determine the apparent rate constant $(k_{\rm obs})$ for the exchange of polymer chains between their micelles. $k_{\rm obs}$ for the mixture of a Ran-D50-Py-D (DP = 172, $N_{\rm agg}$ = 2) micelle and a Ran-D50-Np-D (DP = 170, N_{agg} = 2) micelle at 35 °C (3.0 \times 10⁻² min⁻¹) was almost the same as that for a Ran-**D40-Py-D** (DP = 91, N_{agg} = 2) micelle and a **Ran-D40-Np-D** (DP = 82, $N_{\rm agg}$ = 2) micelle (3.0 × 10⁻² min⁻¹), indicating that the 10 mol% difference in copolymer composition had little effect on the apparent exchange rate of polymer chains. In contrast, $k_{
m obs}$ for the mixture of a Ran-D50-Py-F (DP = 68, N_{agg} = 5) micelle and a **Ran-D50-Np-F** (DP = 38, N_{agg} = 8) micelle at 35 °C (4.6 × 10⁻² min⁻¹) was larger than that for the mixtures of micelles with N_{agg} = 2. This means that the exchange rate of polymer chains increased with increasing N_{agg};⁴² namely, small DP polymers are release/insertion, fragmentation of micelles, and micelle/micelle collision processes. 24,27,43 Since CMC of the random copolymers was independent of DP, faster chain exchange of micelles with large $N_{\rm agg}$ may be promoted by fragmentation and/or collision processes of their micelles.

The $k_{\rm obs}$ values were applied to Arrhenius plots (Figure 7c). Activation energy (Ea) for the chain exchange processes of Ran-**D50-Py-D** ($N_{agg} = 2$)/Ran-D50-Np-D ($N_{agg} = 2$) micelles, Ran-D40-**Py-D** ($N_{\text{agg}} = 2$)/Ran-D40-Np-D ($N_{\text{agg}} = 2$) micelles, and Ran-D50-**Py-F** ($N_{agg} = 5$)/**Ran-D50-Np-F** ($N_{agg} = 8$) micelles were estimated to be 35, 47, and 43 kJ/mol, respectively. E_a values were almost independent of the composition and DP of their copolymers but significantly lower than Ea for the chain exchange of methacrylate-based PEGMA/DMA (50/50 mol%) random copolymer micelles carrying identical PEG chains and dodecyl groups (141 kJ/mol, Figure 1b).53 This importantly demonstrate that the flexible polyacrylate backbones easily induce the exchange of polymer chains between micelles with less temperature effects although rigid polymethacrylate backbones have a higher energy barrier for the exchange of polymer chains. In fact, the acrylate-random copolymer micelles induced chain exchange even at 10 °C (Figure 7), whereas the methacrylate-random copolymer counterparts

required a temperature at least above 25 °C for efficient chain exchange.⁵³ We thus revealed that the flexible acrylate random copolymers were effective for the design of size- and aggregation number-controlled micelles with low CMC yet dynamic chain exchange properties even at low temperature.

Conclusions

In summary, we investigated the self-assembly of acrylatebased amphiphilic random copolymers into micelles in water, focusing on the micelle size and aggregation number, CMC, and dynamic exchange behaviour of polymer chains. Multichain micelles with almost constant size (M_w and R_h) were formed below a threshold DP_{th} and unimer micelles were mainly formed above the DP_{th} . The constant M_w and DP_{th} for the multichain micelles increased with the content of hydrophobic dodecyl groups. CMC of the random copolymers in water was approximately 1×10^{-3} mg/mL regardless of DP and monomer sequence, indicative of the stable formation of micelles at low concentration. Uniquely, the flexible acrylate random copolymer micelles induced the exchange of polymer chains in water even at a low temperature such as 10 °C, though relatively rigid methacrylate random copolymers required at least 25 °C for efficient chain exchange. The chain exchange was promoted by increasing temperature and N_{agg} . Thus, selfassembly of flexible acrylate-based random copolymers in water is suitable for the design of size-controlled micelles with dynamic chain exchange properties even at low temperature; such dynamic yet precise self-assemblies would be useful as encapsulation/release materials or delivery vessels for various applications.

Conflicts of interest

There are no conflicts to declare.

Data availability

The data supporting this study have been included as part of the ESI. Experimental details for the synthesis and characterization of polymers, SEC, NMR, DLS, and fluorescence spectroscopy.

Acknowledgements

This work was supported by Japan Society for the Promotion of Science KAKENHI Grants (JP19K22218, JP20H02787, JP20H05219, JP22H04539, JP23H02008, JP23K26701), JST PRESTO Grant Number JPMJPR24M6, Ogasawara Foundation for the Promotion of Science & Engineering, the Noguchi Institute, the Institute for Chemical Fibers, Japan, Iketani Science and Technology Foundation, Kurita Water and Environment Foundation.

Notes and references

1 L. Zhang and A. Eisenberg, *Science.*, 1995, **268**, 1728–1731.

- B. M. Discher, Y. Y. Won, D. S. Ege, J. C. M. Lee, F. S. Bates, D. E. Discher and D. A. Hammer, *Science.*, 1999, 284, 1143–1146.
- 3 D. E. Discher and A. Eisenberg, *Science.*, 2002, **297**, 967–973.
- 4 Z. Li, E. Kesselman, Y. Talmon, M. A. Hillmyer and T. P. Lodge, *Science.*, 2004, **306**, 98–101.
- 5 R. K. O'reilly, C. J. Hawker and K. L. Wooley, *Chem. Soc. Rev.*, 2006, **35**, 1068–1083.
- 6 A. O. Moughton, M. A. Hillmyer and T. P. Lodge, *Macromolecules*, 2012, **45**, 2–19.
- 7 Y. Mai and A. Eisenberg, *Chem. Soc. Rev.*, 2012, **41**, 5969–5985.
- Y. Lu, J. Lin, L. Wang, L. Zhang and C. Cai, *Chem. Rev.*, 2020, 120, 4111–4140.
- 9 T. Terashima, Polym. J., 2014, 46, 664-673.
- S. Mavila, I. Rozenberg and N. G. Lemcoff, *Chem. Sci.*, 2014, 5, 4196–4203.
- 11 M. Gonzalez-Burgos, A. Latorre-Sanchez and J. A. Pomposo, *Chem. Soc. Rev.*, 2015, **44**, 6122–6142.
- 12 P. W. Roesky and C. Barner-kowollik, *Polym. Chem.*, 2015, 6, 4358–4365.
- 13 C. Song, L. Li, L. Dai and S. Thayumanavan, *Polym. Chem.*, 2015, **6**, 4828–4834.
- 14 A. M. Hanlon, C. K. Lyon and E. B. Berda, *Macromolecules*, 2016, **49**, 2–14.
- 15 S. Mavila, O. Eivgi, I. Berkovich and N. G. Lemcoff, *Chem. Rev.*, 2016, **116**, 878–961.
- 16 O. Altintas and C. Barner-Kowollik, *Macromol. Rapid Commun.*, 2016, **37**, 29–46.
- 17 E. S. Gil and S. M. Hudson, *Prog. Polym. Sci.*, 2004, **29**, 1173–1222.
- 18 F. D. Jochum and P. Theato, *Chem. Soc. Rev.*, 2013, **42**, 7468–7483.
- 19 M. I. Gibson and R. K. O'reilly, Chem. Soc. Rev., 2013, 42, 7204–7213.
- 20 A. P. Esser-Kahn, S. A. Odom, N. R. Sottos, S. R. White and J. S. Moore, *Macromolecules*, 2011, 44, 5539–5553.
- 21 Z. Jiang, H. Liu, H. He, A. E. Ribbe and S. Thayumanavan, *Macromolecules*, 2020, **53**, 2713–2723.
- 22 K. Kataoka, A. Harada and Y. Nagasaki, *Adv. Drug Deliv. Rev.*, 2012, **64**, 37–48.
- 23 N. Kamaly, B. Yameen, J. Wu and O. C. Farokhzad, *Chem. Rev.*, 2016, **116**, 2602–2663.
- 24 Y. Wang, C. M. Kausch, M. Chun, R. P. Quirk and W. L. Mattice, *Macromolecules*, 1995, **28**, 904–911.
- P. Bhargava, J. X. Zheng, P. Li, R. P. Quirk, F. W. Harris and S. Z.
 D. Cheng, *Macromolecules*, 2006, 39, 4880–4888.
- 26 R. Lund, L. Willner, J. Stellbrink, P. Lindner and D. Richter, *Phys. Rev. Lett.*, 2006, **96**, 1–4.
- 27 T. P. Lodge, C. L. Seitzinger, S. C. Seeger, S. Yang, S. Gupta and K. D. Dorfman, *ACS Polym. Au*, 2022, **2**, 397–416.
- 28 H. Yamamoto and Y. Morishima, *Macromolecules*, 1999, **32**, 7469–7475.
- 29 H. Yamamoto, I. Tomatsu, A. Hashidzume and Y. Morishima, Macromolecules, 2000, 33, 7852–7861.
- 30 S. Yusa, A. Sakakibara, T. Yamamoto and Y. Morishima, *Macromolecules*, 2002, **35**, 10182–10188.
- 31 T. Kawata, A. Hashidzume and T. Sato, *Macromolecules*, 2007, **40**, 1174–1180.
- 32 K. Dan, N. Bose and S. Ghosh, *Chem. Commun.*, 2011, **47**, 12491–12493.
- 33 L. Li, K. Raghupathi, C. Song, P. Prasad and S. Thayumanavan, *Chem. Commun.*, 2014, **50**, 13417–13432.
- 34 R. Nakahata and S. I. Yusa, *Langmuir*, 2019, **35**, 1690–1698.
- 35 Y. Hirai, T. Terashima, M. Takenaka and M. Sawamoto, *Macromolecules*, 2016, **49**, 5084–5091.
- 36 G. Hattori, Y. Hirai, M. Sawamoto and T. Terashima, *Polym. Chem.*, 2017, 8, 7248–7259.

- 37 G. Hattori, M. Takenaka, M. Sawamoto and T. Terashima, *J. Am. Chem. Soc.*, 2018, **140**, 8376–8379.
- 38 Y. Kimura, T. Terashima and M. Sawamoto, *Macromol. Chem. Phys.*, 2017, **218**, 1700230.
- 39 S. Imai, Y. Hirai, C. Nagao, M. Sawamoto and T. Terashima, *Macromolecules*, 2018, **51**, 398–409.
- 40 M. Shibata, M. Matsumoto, Y. Hirai, M. Takenaka, M. Sawamoto and T. Terashima, *Macromolecules*, 2018, **51**, 3738–3745.
- 41 S. Imai, M. Takenaka, M. Sawamoto and T. Terashima, *J. Am. Chem. Soc.*, 2019, **141**, 511–519.
- 42 M. Hibino, K. Tanaka, M. Ouchi, T. Terashima, *Macromolecules*, 2022, **55**, 178-189.
- 43 M. Hibino, S. I. Takata, K. Hiroi, H. Aoki and T. Terashima, *Macromolecules*, 2023, **56**, 2955–2964.
- 44 D. Taura, A. Hashidzume, Y. Okumura and A. Harada, *Macromolecules*, 2008, **41**, 3640–3645.
- 45 D. Taura, Y. Taniguchi, A. Hashidzume and A. Harada, *Macromol. Rapid Commun.*, 2009, **30**, 1741–1744.
- 46 M. Ueda, A. Hashidzume and T. Sato, *Macromolecules*, 2011, 44, 2970–2977.
- 47 K. Uramoto, R. Takahashi, K. Terao and T. Sato, *Polym. J.*, 2016, **48**, 863–867.
- 48 H. Kono, M. Hibino, D. Ida, M. Ouchi and T. Terashima, *Macromolecules*, 2023, **56**, 6086–6098.
- 49 K. Kalyanasundaram and J. K. Thomas, *J. Am. Chem. Soc.*, 1977, **99**, 2039–2044.
- 50 M. Wilhelm, C. Le Zhao, Y. Wang, R. Xu, M. A. Winnik, J. L. Mura, G. Riess and M. D. Croucher, *Macromolecules*, 1991, 24, 1033–1040.
- 51 P. Raffa, D. A. Z. Wever, F. Picchioni and A. A. Broekhuis, *Chem. Rev.*, 2015, **115**, 8504–8563.
- 52 A. Goswami, G. Verma, P. A. Hassan and S. S. Bhagwat, *J. Dispers. Sci. Technol.*, 2015, **36**, 885–891.
- 53 R. Kanno, H. Kono, A. Ryoki, M. Ouchi, T. Terashima, *J. Am. Chem. Soc.*, 2024, **146**, 30848–30859.

Data availability statements

Precisely Controlled yet Dynamically Exchanged Micelles via the Self-Assembly of Amphiphilic Acrylate Random Copolymers in Water

Hiroyuki Kono, Makoto Ouchi, and Takaya Terashima

Data availability:

The data supporting this study have been included as part of the Supplementary Information. In the ESI, experimental details for the synthesis and characterization of polymers and polymer micelles and the data of SEC, NMR, DLS, and fluorescence spectroscopy are shown.