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# High efficiency coupled electrocatalytic CO<sub>2</sub> reduction to C<sub>2</sub>H<sub>4</sub> with 5-hydroxymethylfurfural oxidation over Cu-based nanoflower electrocatalysts†

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To improve the total value of the redox products and energy conversion efficiency in CO<sub>2</sub> electroreduction (CO<sub>2</sub>RR) and 5-hydroxymethylfurfural electrooxidation (HMFOR), commercial Cu foam was chemically oxidized into nanoflower CuO on its surface and used as an electrocatalyst in both CO<sub>2</sub>RR and HMFOR. In CO<sub>2</sub>RR, the prepared CuO nanoflower on Cu foam (CuO-NF@Cu) was quickly reduced to hybrid Cu<sub>2</sub>O/Cu nanoflower (Cu<sub>2</sub>O/Cu-NF@Cu). At a reduction potential of −0.95 V (vs. RHE), a high C<sub>2</sub>H<sub>4</sub> faradaic efficiency (FE) up to 70% with a current density of 104.5 mA cm<sup>−2</sup> can be obtained within 45 h of testing. In HMFOR, CuO-NF@Cu gave a 99.3% FDCA FE at a potential of 1.62 V (vs. RHE). Moreover, CO<sub>2</sub>RR and HMFOR can be coupled together with CuO-NF@Cu as the anode electrocatalyst and Cu<sub>2</sub>O/Cu-NF@GDL as the cathode electrocatalyst. A current density up to 188.8 mA cm<sup>−2</sup> at a cell voltage of 2.75 V can be obtained. FEs of FDCA and C<sub>2</sub>H<sub>4</sub> up to 96.6%/74.5%, respectively, were achieved in coupled CO<sub>2</sub>RR-HMFOR within 5 h. This work makes it easy to simultaneously efficiently convert CO<sub>2</sub> to high-value C<sub>2</sub>H<sub>4</sub> and upgrade a renewable biomass platform compound.

## Introduction

The rapid development of industry over the past hundred years has led to the large-scale use of nonrenewable fossil resources, causing a substantial increase in the CO<sub>2</sub> concentration in the atmosphere.<sup>1–4</sup> Developing and utilizing renewable energy and resources, such as solar, wind, electric, and biomass resources, and selectively reducing CO<sub>2</sub> generated by fossil resources to valuable fuels or chemicals with clean and renewable energy is

significant for future development.<sup>5–7</sup> The selective reduction of CO<sub>2</sub> (CO<sub>2</sub>RR) into one specific multi-carbon product with high Faraday efficiency (FE) over one easily prepared electrocatalyst is far from practical application owing to the uncertain and kinetically sluggish C–C coupling and H addition.<sup>8–12</sup> Moreover, a sluggish oxygen evolution reaction (OER) at the anode results in high anodic potential and as high as 94.5% of the input energy is consumed at the anode with low-value O<sub>2</sub> produced.<sup>13–15</sup> Thus, substituting OER with reactions of lower energy requirements and value-added oxidative products would significantly reduce the anodic potentials, as well as raise the efficiency for CO<sub>2</sub>RR.<sup>16–18</sup>

5-Hydroxymethylfurfural (HMF) is an important biomass-derived platform molecule. It can be oxidized into 2,5-furandicarboxylic acid (FDCA) under certain conditions.<sup>19</sup> FDCA has a similar molecular structure to terephthalic acid and can be used as an alternative chemical in the fields of fine chemicals and medicines.<sup>20</sup> Traditionally, the production of FDCA from HMF needs harsh reaction conditions, such as a high reaction temperature, high oxygen pressure and noble metal catalysts.<sup>21,22</sup> Electrocatalysis of HMF (HMFOR) can be conducted under mild conditions without high reaction temperature, high oxygen pressure, or an expensive catalyst.<sup>23</sup> However, the reported electrocatalysts for HMF oxidation are prepared with complicated procedures or give low FDCA selectivity.<sup>24–26</sup> On the other hand, the kinetics of the synergistic hydrogen evolution reaction (HER) at the cathode is not ideal and restricts the anode HMF oxidation. Thus, coupling CO<sub>2</sub>RR and HMFOR seems an attractive approach to solving the above problem in CO<sub>2</sub>RR and HMFOR. In this field, researchers have tried coupling CO<sub>2</sub>RR (to formic acid, or CO) with HMFOR (to FDCA, maleic acid, formic acid),<sup>18,27,28</sup> CO<sub>2</sub>RR (to CO or formate) with CH<sub>3</sub>OH oxidation (to formate),<sup>5,29</sup> HMFOR (to FDCA) with 4-nitrophenol hydrogenation,<sup>30</sup> etc. Generally, only a C<sub>1</sub> product was achieved in the CO<sub>2</sub>RR coupled reaction and there was low FDCA selectivity in the HMFOR coupled reaction with low current density and complicated, different electrocatalyst preparation procedures in the anode and cathode.<sup>31</sup> Therefore, an

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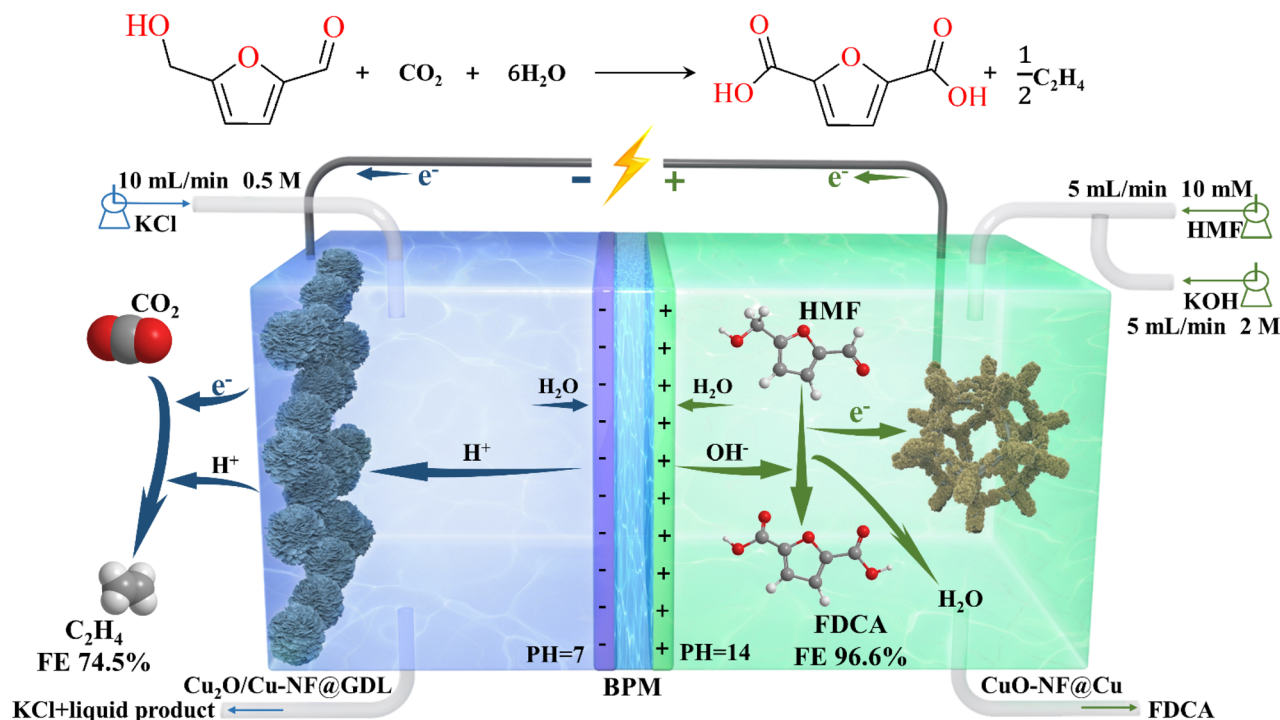
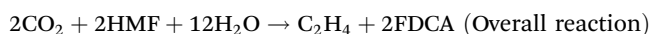
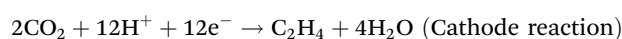


Fig. 1 Schematic illustration of integrated electrolysis flow cell coupling CO<sub>2</sub>RR with HMFOR.

easily prepared electrocatalyst that can simultaneously give high FDCA selectivity in anode HMF oxidation and high FE in cathode CO<sub>2</sub>RR to produce high-value C<sub>2</sub> products is very important for the utilization of renewable biomass resources, renewable electric energy, and cyclic utilization of CO<sub>2</sub> resources.

In the past decades, Au, Pt, Pd, Ru or their alloys have been widely studied as anode electrocatalysts in HMFOR.<sup>32–34</sup> Recently, studies on transition metal (Fe, Co, Ni, Cu, *etc.*) – based electrocatalysts have caused rapid development in HMFOR and even higher target product selectivity and energy efficiency have been obtained.<sup>35–39</sup> In general, researchers concluded that the active phases in HMF electrooxidation with transition metal catalysts should be their nitride or oxide compounds with unique nanostructures. The atomic vacancies and local structural disorder created by exotic species like oxygen,<sup>37,40</sup> nitride,<sup>38</sup> sulphur,<sup>39,41</sup> phosphorus,<sup>36</sup> and selenium,<sup>42</sup> the morphology of the nanostructure, and the valence states of the metal sites are responsible for their activity in HMFOR. In CO<sub>2</sub>RR, a copper-based catalyst was shown to be an efficient electrocatalyst toward C<sub>2</sub>+ products.<sup>43–46</sup> Defect sites with low-coordinated Cu atoms and polarized Cu<sup>δ+</sup> (0 < δ ≤ 1) were deemed responsible for promoting the C–C coupling step at low overpotentials in CO<sub>2</sub>RR.<sup>44</sup> This feature indicates that one elaborately prepared oxide-derived Cu catalyst may be suitable in both CO<sub>2</sub>RR and HMFOR as follows.



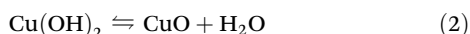
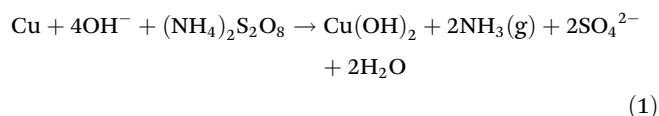
Here, a simply prepared CuO nanoflower (CuO-NF) sheet over a Cu foam surface (CuO-NF@Cu) was prepared and used as the electrocatalyst in CO<sub>2</sub>RR and HMFOR. In CO<sub>2</sub>RR, the CuO-NF in the resultant CuO-NF@Cu quickly converted into the Cu<sub>2</sub>O phase in CO<sub>2</sub>RR and formed a stable Cu<sup>+</sup>/Cu interface. The obtained Cu<sub>2</sub>O/Cu-NF@Cu exhibited facilitated kinetics of CO<sub>2</sub>RR into C<sub>2</sub>H<sub>4</sub>, leading to a high C<sub>2</sub>H<sub>4</sub> FE of *ca.* 70% at –0.95 V (*vs.* RHE), along with a current density of 104.5 mA cm<sup>–2</sup> within the tested 45 h. On the other side, CuO-NF@Cu can give 100% HMF conversion with a 99.6% FDCA yield and 99.3% FDCA FE at a potential of 1.62 V (*vs.* RHE) in 45 min. Moreover, CO<sub>2</sub>RR and HMFOR can be coupled together in an integrated flowing cell with CuO-NF@Cu as the electrocatalyst in both the anode and cathode (Fig. 1). A stable current density up to 188.8 mA cm<sup>–2</sup> at a cell voltage of 2.75 V can be obtained, along with a FDCA FE up to 96.6% and a C<sub>2</sub>H<sub>4</sub> FE up to 74.5%. The excellent performance of the Cu-based nanoflower electrocatalyst demonstrates great promise for the large scale cyclic utilization of CO<sub>2</sub> resources generated from coal or oil and biomass valorization, facilitated by renewable electric energy from wind, solar, *etc.*

## Results and discussion

### Catalyst preparation and characterization

In this study, Cu nanoflower sheet catalysts were grown over foam Cu. Detailed catalyst preparation methods are given in the ESI† and are illustrated in Fig. S1 (in the ESI†). The mor-

phology and microstructure of the catalyst in different stages in Fig. S1† are characterized by SEM and shown in Fig. S2.† Without stirring, the yellow surface of the pristine Cu foam turned blue in 10 min, indicating the formation of  $\text{Cu}(\text{OH})_2$  over the Cu foam. The SEM images in Fig. S2a and b† show that the surface of the Cu foam turned from smooth to rough and  $\text{Cu}(\text{OH})_2$  nanorods uniformly covered the surface of the Cu foam ( $\text{Cu}(\text{OH})_2\text{-NR@Cu}$ ). After 20 min, the surface turned dark blue, indicating that partial  $\text{Cu}(\text{OH})_2$  had dehydrated to CuO. The SEM images showed that CuO nucleated and formed a small amount of CuO nanoflowers over the  $\text{Cu}(\text{OH})_2$  nanorods ( $\text{CuO/Cu}(\text{OH})_2\text{-NR@Cu}$ , Fig. S2c†). Vigorous stirring in the catalyst preparation can accelerate this transformation and a black surface can be found in 20 min with stirring in the catalyst preparation, implying that a majority of the Cu foam surface is covered by CuO. The SEM image in Fig. S2d† shows that CuO nanoflowers covered the whole surface of the Cu foam ( $\text{CuO-NF@Cu}$ ). Under alkaline conditions, the inorganic polymerization reactions of  $\text{Cu}(\text{OH})_2$ ,  $\text{CuO/Cu}(\text{OH})_2$ , and CuO are as follows.



At first, Cu atoms on the Cu foam surface dissolved into  $\text{Cu}^{2+}$  in the presence of  $(\text{NH}_4)_2\text{S}_2\text{O}_8$ . In the presence of  $\text{OH}^-$ , partial  $\text{Cu}^{2+}$  formed  $\text{Cu}(\text{OH})_2$  and deposited on the Cu foam surface ( $\text{Cu}(\text{OH})_2\text{@Cu}$ , Fig. S2b†). With the reaction time prolonged to 20 min and  $\text{Cu}^{2+}$  saturated in the solution,  $\text{Cu}(\text{OH})_2$  dehydrated to CuO, which is thermodynamically favorable. The formed CuO nucleated and grew on the surface with the production and dehydration of  $\text{Cu}(\text{OH})_2$ . Depending on the rates of formation and dehydration of  $\text{Cu}(\text{OH})_2$ , a mixture of CuO and  $\text{Cu}(\text{OH})_2$  can exist on the surface ( $\text{CuO/Cu}(\text{OH})_2\text{@Cu}$ , Fig. S2c†). Stirring during the reaction accelerated the nucleation rate until the surface oxygen content no longer increased, indicating that the surface was completely covered by the oxide layer (Fig. S2d†). Finally, vertically arranged and densely stacked CuO nanoflowers ( $\text{CuO-NF@Cu}$ ) were synthesized. This method not only prevents the increase of contact resistance between the nanorods/flowers and the conductive substrate but also interferes with the diffusion of the electrolyte, thereby improving the catalytic performance.

Further SEM observations and analysis of the  $\text{CuO-NF@Cu}$  were conducted and are shown in Fig. 2. An EDS elemental analysis of the nanoflowers and the EDS elemental spectra showed that the Cu and O elements are evenly distributed on the nanoflowers, with an atomic ratio of oxygen to copper close to 1:1, indicating the successful preparation of  $\text{CuO-NF@Cu}$  (Fig. 2b and c). Meanwhile, an EDS elemental analysis of the nanoflowers on  $\text{CuO/Cu}(\text{OH})_2\text{@Cu}$  showed that the atomic ratio of oxygen to copper was also close to 1:1 (Fig. S3†), confirming that CuO nanoflowers nucleated and

grew on the  $\text{Cu}(\text{OH})_2$  nanorods. High-magnification SEM and HRTEM in Fig. 3a and b indicate that the CuO nanoflowers are composed of ultra-thin CuO nanosheets, with an average width of *ca.* 200 nm. The thickness of the nanosheets was measured with atomic force microscopy (AFM) and found to be *ca.* 4.5 nm (Fig. 3c). The ultra-thin nanosheets of  $\text{CuO-NF@Cu}$  provide more catalytic active sites during the electrocatalytic process. The above characterization demonstrated the successive preparation of CuO nanoflowers with nanosheets over Cu foam.

The crystal structure of the catalysts was determined by X-ray diffraction (XRD) and is shown in Fig. S4.† All the catalysts exhibit three high-intensity diffraction peaks at  $43.316^\circ$ ,  $50.448^\circ$ , and  $74.142^\circ$ , corresponding to the Cu (111), (200), and (220) planes (Fig. S4a†), respectively (JCPDS Card No. 04-0836), indicating that  $\text{Cu}(\text{OH})_2$  nanorods and CuO nanoflowers directly grew on the surface of the Cu foam (Fig. S4b–d†). The XRD pattern of  $\text{Cu}(\text{OH})_2\text{-NR@Cu}$  in Fig. S4b† exhibits typical diffraction peaks at  $34.061^\circ$ ,  $39.800^\circ$  and  $53.210^\circ$  which are respectively assigned to the (002), (130) and (150) planes of the orthorhombic crystal phase of  $\text{Cu}(\text{OH})_2$  (JCPDS Card No. 13-0420). The XRD pattern of  $\text{CuO/Cu}(\text{OH})_2\text{@Cu}$  in Fig. S4c† shows typical diffraction peaks at  $36.636^\circ$  and  $42.558^\circ$ , which correspond to the (002) and (111) planes of the orthorhombic crystal phase of CuO (JCPDS Card No. 80-1917), in addition to the diffraction peaks assigned to  $\text{Cu}(\text{OH})_2$ . For the  $\text{CuO-NF@Cu}$  catalyst, no other phases were observed except for Cu (JCPDS Card No. 04-0836) and CuO (JCPDS Card No. 80-1917), as shown in Fig. S4d.†

To further investigate the surface chemical composition and valence state of the catalyst, XPS analysis was performed and is given in Fig. S5.† Fig. S5a† shows the XPS spectrum of the  $\text{CuO-NF@Cu}$  catalyst, in which the characteristic peaks of copper, oxygen, and carbon can be clearly observed. According to the XPS quantitative analysis of the  $\text{CuO-NF@Cu}$  catalyst, the ratio of oxygen to copper in the catalyst is 1.15:1, which is almost consistent with the ratio of 1.08:1 measured by inductively coupled plasma optical emission spectroscopy (ICP-OES). In the high-resolution Cu 2p spectrum of the catalyst, the two peaks of Cu 2p<sub>1/2</sub> and Cu 2p<sub>3/2</sub> are located at 952.90 eV and 933.11 eV, respectively (Fig. S5b,† bottom), and are caused by elemental copper. The strong shake-up peaks at 943.65 eV and 963.00 eV indicate the presence of  $\text{Cu}^{2+}$  (Fig. S5b,† top) and cannot be found in the spectrum of the copper foam. The main peak of Cu 2p<sub>3/2</sub> can be further divided into two peaks, those of  $\text{Cu}^{2+}$  from CuO at 935.47 eV and  $\text{Cu}^0$  at 933.82 eV, because of the inevitable minor amount of Cu remaining unreacted in the process (Fig. S5c†). In addition, the O 1s spectrum (Fig. S5d†) can be divided into two peaks, which belong to the Cu–O bond of CuO at 529.44 eV and a peak at 531.26 eV attributed to hydroxyl groups adsorbed on the surface.

#### Cathodic electrochemical $\text{CO}_2$ reduction to ethylene

$\text{CuO-NF@Cu}$  was used as the working electrode for electrochemical  $\text{CO}_2$  reduction, with an Ag/AgCl electrode as the



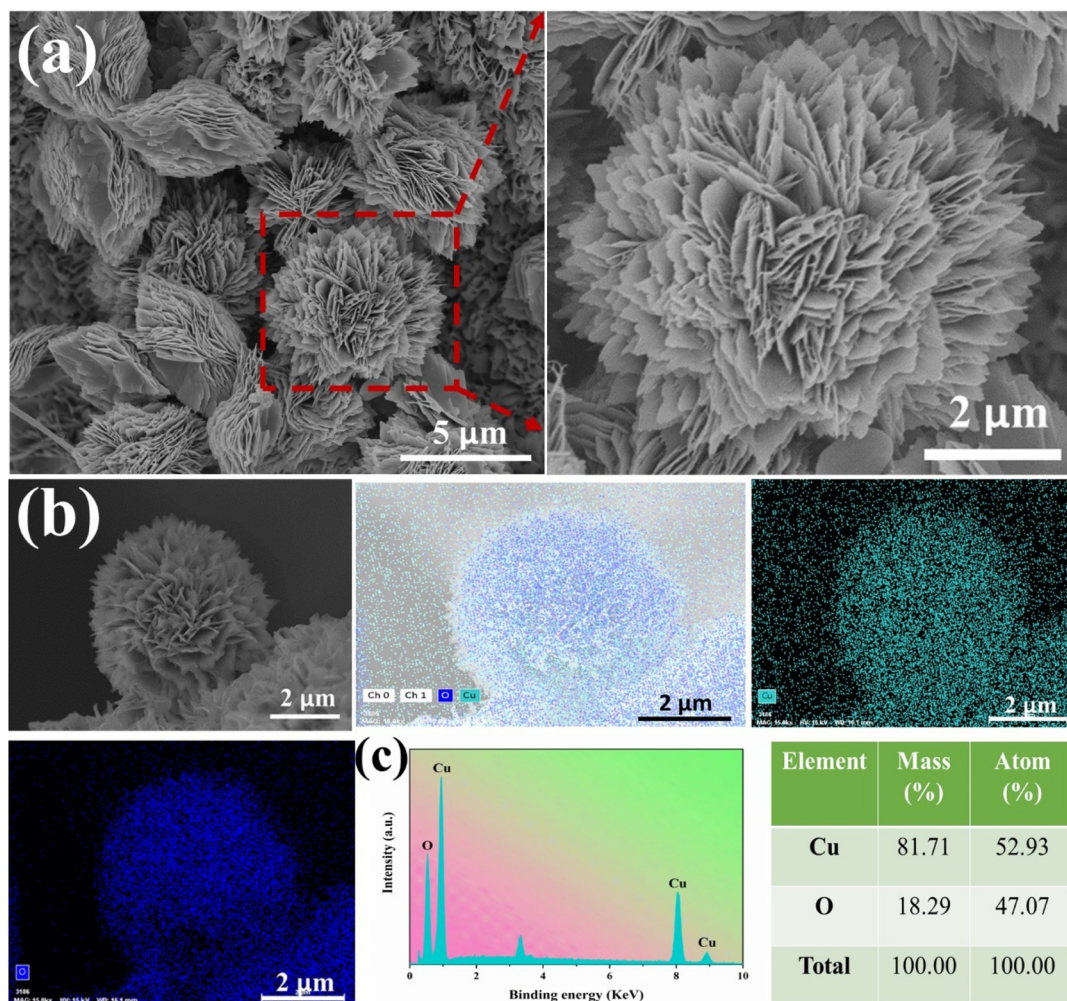


Fig. 2 (a) SEM images of CuO-NF@Cu, (b) elemental mapping images of CuO-NF@Cu, (c) EDS spectrum of CuO-NF@Cu.

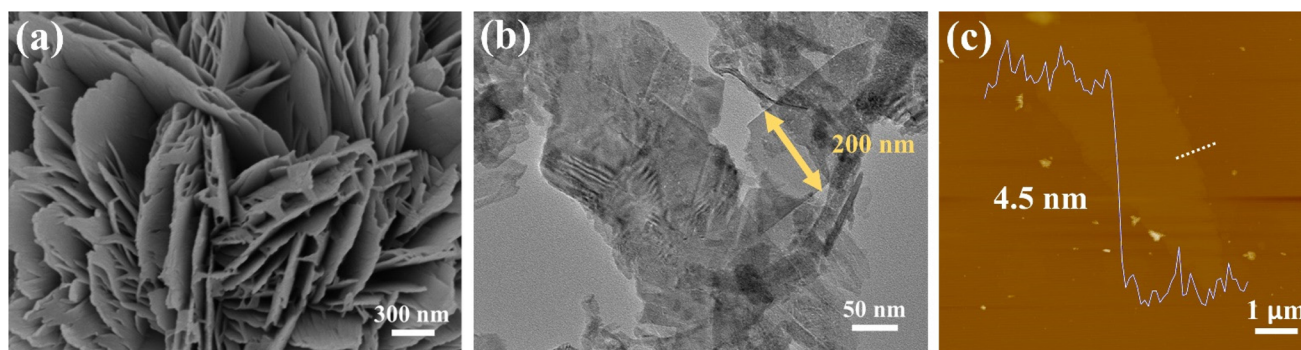
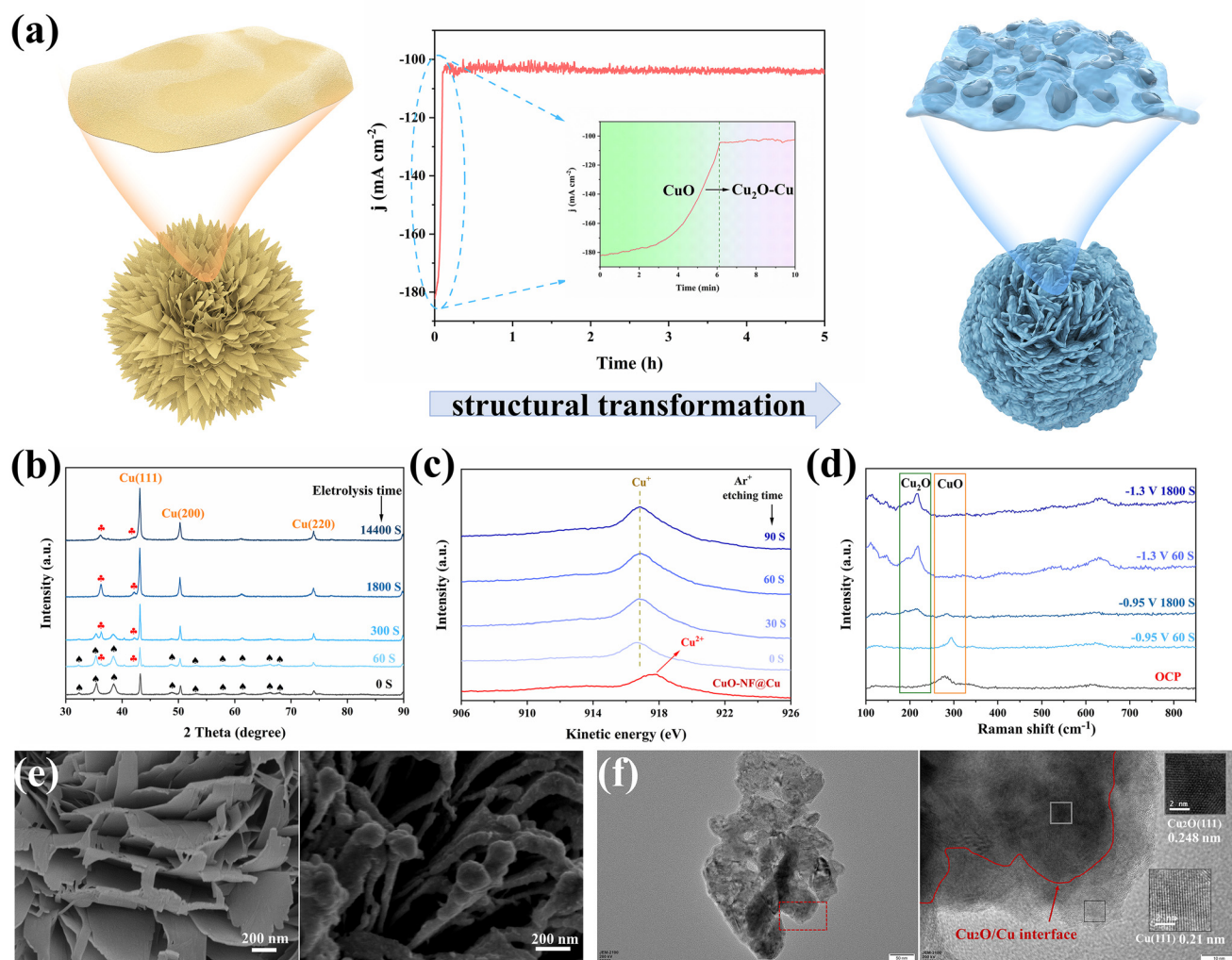


Fig. 3 (a) High-magnification SEM and (b) HRTEM images of CuO-NF@Cu at different magnifications; the CuO flakes are about 200 nm in width and 6 nm in thickness. (c) AFM image of the CuO-NF@Cu and the corresponding height profile from the dashed line.

reference electrode and a platinum foil as the counter electrode. *In situ* constant current reduction was performed at  $-0.95$  V (vs. RHE) under a flowing  $\text{CO}_2$  atmosphere. The current decreased sharply in the first 6 min (360 seconds) of electrochemical reduction from *ca.*  $182.0 \text{ mA cm}^{-2}$  to *ca.*

$104.5 \text{ mA cm}^{-2}$  (Fig. 4a, middle) and the black surface of the CuO-NF@Cu catalyst quickly turned reddish-brown (Fig. S6†), indicating that CuO in CuO-NF@Cu can be quickly reduced. To check the composition of the reduced CuO-NF@Cu, X-ray diffraction (XRD) over CuO-NF@Cu and reduced CuO-NF@Cu



**Fig. 4** (a) Schematic illustration of morphology changes during the electrochemical reduction of CuO-NF@Cu, (b) XRD patterns of CuO-NF@Cu at different reduction times, (c) Cu LMM Auger spectra of CuO-NF@Cu and Cu<sub>2</sub>O/Cu-NF@Cu with different Ar<sup>+</sup> etching times, (d) *in situ* Raman spectra of Cu<sub>2</sub>O/Cu-NF@Cu in different conditions, (e) SEM images of fresh CuO-NF@Cu and Cu<sub>2</sub>O/Cu-NF@Cu, (f) TEM and HRTEM images of Cu<sub>2</sub>O/Cu-NF@Cu.

at different reduction times was conducted and the results are shown in Fig. 4b. The XRD pattern indicates that, besides Cu diffraction peaks, only CuO diffraction peaks were observed in CuO-NF@Cu (spade marks in Fig. 4b). After 60 seconds of electrochemical reduction, diffraction peaks attributed to Cu<sub>2</sub>O begin to appear (red club marks in Fig. 4b). After 300 seconds of electrochemical reduction, the CuO peak completely disappears, while the Cu<sub>2</sub>O peak still exists, even if the reduction time is extended to 1800 seconds (0.5 h) and 14 400 seconds (4 h). These results implied that surface CuO in CuO-NF@Cu can be reduced into Cu<sub>2</sub>O which is stable under current reduction conditions. Thus, the cathode catalyst is named Cu<sub>2</sub>O/Cu-NF@Cu in the following text.

To check the surface and subsurface Cu valence states of CuO-NF@Cu and Cu<sub>2</sub>O/Cu-NF@Cu, X-ray photoelectron spectroscopy (XPS) and Auger electron spectroscopy (AES) were conducted with different times of Ar<sup>+</sup> etching over Cu<sub>2</sub>O/Cu-

NF@Cu, as shown in Fig. 4c and Fig. S7, S8†. The Cu<sub>2</sub>O/Cu-NF@Cu sample was obtained by 2 hours of electrochemical reduction under the above-mentioned conditions to ensure that its structure was the same as the cathode used in the long-time CO<sub>2</sub>RR. In CuO-NF@Cu, the Cu LMM Auger peak at 917.4 eV and the typical satellite peaks can be attributed to Cu<sup>2+</sup> species. In the XPS spectra of Cu<sub>2</sub>O/Cu-NF@Cu and Ar<sup>+</sup> etched Cu<sub>2</sub>O/Cu-NF@Cu with etching times from 0–90 seconds, no Cu<sup>2+</sup> but only Cu<sup>+</sup> peaks were found, indicating that the original Cu<sup>2+</sup> species in CuO-NF@Cu were completely reduced to Cu<sup>+</sup> after electrolysis. The Cu LMM spectrum shows that Cu<sup>+</sup> species can exist in Cu<sub>2</sub>O/Cu-NF@Cu even after 90 s of Ar<sup>+</sup> etching (estimated etching depth of 30 nm), which can completely exclude the surface oxide layer formed by air oxidation (usually <5 nm). Meanwhile, the peaks belonging to the Cu and O elements are compared in Fig. S7 and S8†: the Cu peak intensity increased and the O peak intensity decreased



with detection depth increasing, indicating that the original CuO phase was generated over the Cu foam surface. Further *in situ* Raman spectroscopy was performed to identify the valence of Cu during electrolysis (Fig. 4d). The peaks of Raman shifts at 287, 337, and 622  $\text{cm}^{-1}$  belong to CuO under open circuit potential (OCP) and can be clearly observed. When Cu<sub>2</sub>O/Cu-NF@Cu was reduced at  $-0.95$  V (vs. RHE) for 60 s, the characteristic peak belonging to CuO became weakened and the peak belonging to Cu<sub>2</sub>O began to appear.<sup>47</sup> After 1800 s, the CuO peak completely disappeared but the Cu<sub>2</sub>O peak remained. At higher potential ( $-1.3$  V vs. RHE in Fig. 4d), Cu<sup>+</sup> can also be retained even after undergoing long-time electrochemical reduction. Therefore, it can be concluded that Cu<sup>+</sup> species will be preserved during the electrochemical CO<sub>2</sub>RR process.

Morphology changes before and after electrochemical reduction were characterized by SEM and TEM. As shown in Fig. 4e and Fig. S9,† SEM images of electrochemically reduced CuO-NF@Cu indicate that the vertically arranged and densely stacked laminated nanostructures are retained. However, the flakes are no longer composed of CuO nanosheets but nanoparticles. Each ultrathin nanosheet transformed into a thicker nanosheet composed of small nanoparticles.

The HRTEM images provide direct evidence for the self-evolution of nanoflakes in CuO-NF@Cu to nanoparticles in Cu<sub>2</sub>O/Cu-NF@Cu, as shown in Fig. 3b and Fig. S10.† The complete Cu<sub>2</sub>O/Cu-NF@Cu nanoflakes were reduced and restructured to Cu<sub>2</sub>O and Cu, resulting in a large number of grain boundary interfaces (GBs) between Cu<sub>2</sub>O and Cu (Fig. 4f). The large Cu particles were wrapped by Cu<sub>2</sub>O facets and a Cu (111) plane with a lattice spacing of 0.21 nm was observed on the larger Cu particles, while a Cu<sub>2</sub>O (111) plane with a lattice spacing of 0.248 nm was observed around the Cu particles. As a result, a large number of Cu<sub>2</sub>O (111) facets and Cu (111) particles provide rich Cu<sub>2</sub>O (111)/Cu (111) (Cu<sup>+</sup>/Cu<sup>0</sup>) interfaces in the Cu<sub>2</sub>O/Cu-NF@Cu nanoflakes.

Electrochemical CO<sub>2</sub> reduction over the Cu<sub>2</sub>O/Cu-NF@Cu electrode was evaluated at atmospheric pressure using a three-electrode H-type electrolytic cell setup. Fig. 5a shows the LSV curves of Cu<sub>2</sub>O/Cu-NF@Cu in 0.5 M KCl water solution saturated with N<sub>2</sub> or CO<sub>2</sub> at a scan rate of 2 mV s<sup>-1</sup> (methods in ESI†). Much higher current was observed in a flowing CO<sub>2</sub> atmosphere than in N<sub>2</sub> atmosphere at all scan potentials, demonstrating the significant CO<sub>2</sub>RR performance of Cu<sub>2</sub>O/Cu-NF@Cu. The reduction products were then detected and the liquid products were mainly ethanol and formate, with a small amount of acetate (Fig. S11†). In the gas products, C<sub>2</sub>H<sub>4</sub> and H<sub>2</sub> were the main reduction products (Fig. S12†). Over Cu<sub>2</sub>O/Cu-NF@Cu at various potentials, the highest Faraday efficiencies (FE) of 70% for C<sub>2</sub>H<sub>4</sub> and 59.6% for C<sub>2</sub>H<sub>4</sub> selectivity were obtained at  $-0.95$  V (Fig. 5b, Fig. S13 and Table S1 in ESI†). Additionally, the FE<sub>H<sub>2</sub></sub> was only 17.5% at  $-0.95$  V over Cu<sub>2</sub>O/Cu-NF@Cu. The high proportion of C<sub>2+</sub>/C<sub>1</sub> (CO, CH<sub>4</sub>, formate) in the CO<sub>2</sub>RR products further confirms the kinetic advantage of C<sub>2</sub>H<sub>4</sub> and C<sub>2+</sub> formation over Cu<sub>2</sub>O/Cu-NF@Cu (Fig. 5c). According to previous research, the performance

observed in stability tests of OD-Cu<sup>48,49</sup> catalysts has consistently failed to meet satisfactory levels. Long-term electrolysis at  $-0.95$  V (vs. RHE) over Cu<sub>2</sub>O/Cu-NF@Cu (Fig. 5d) showed a stable *i*-*t* curve with a current density of 104.5 mA cm<sup>-2</sup> and high C<sub>2</sub>H<sub>4</sub> FE (FE<sub>C<sub>2</sub>H<sub>4</sub></sub> ≈ 70%) over 45 hours.

After long-term testing, XRD characterization of the recovered Cu<sub>2</sub>O/Cu-NF@Cu catalyst indicated that the characteristic peak of Cu<sub>2</sub>O was retained in the spectrum (Fig. S14†). XPS analysis showed that Cu<sub>2</sub>O/Cu-NF@Cu retained a similar Cu/Cu<sup>+</sup> interface structure and Cu<sup>+</sup> content in the subsurface (Fig. S15†). SEM images showed that the nanoflakes were still stacked densely and orderly after 45 hours of electrolysis without any aggregation, and only a slight increase in the thickness of nanoflakes was observed (Fig. S16a and b†). HRTEM images showed that the Cu<sub>2</sub>O/Cu (Cu/Cu<sup>+</sup>) interfaces in the Cu<sub>2</sub>O/Cu-NF@Cu catalyst were preserved (Fig. S16c and d†), indicating the excellent stability of the Cu<sub>2</sub>O/Cu-NF@Cu catalyst compared to previous reported catalytic systems (Table S2†). The stable Cu/Cu<sup>+</sup> interfaces are believed to be active catalytic sites for C<sub>2</sub>H<sub>4</sub> formation and the detailed mechanism can be found in previous reports.<sup>50,51</sup> Moreover, this prepared CuO-NF@Cu is more stable than those previously reported. This work provides a feasible procedure for preparing a stable Cu/Cu<sup>+</sup> interface-containing nanocatalyst and verifies its effectiveness in CO<sub>2</sub>RR and HMFOR.

#### Anodic high efficiency oxidation of HMF to FDCA

To investigate the activity of the prepared CuO-NF@Cu catalyst in anodic HMF electrochemical oxidation, a series of electrochemical tests was conducted in a three-electrode setup. The linear sweep voltammetry (LSV) curves of CuO-NF@Cu in 0.1 mol L<sup>-1</sup> KOH solution with and without HMF are given in Fig. S17.† With HMF, an obviously higher current can be observed than that without HMF under the studied potentials. The copper foam has no effect on the anodic HMF oxidation (Fig. S18†). The electrocatalytic oxidation of HMF to FDCA involves two possible oxidation pathways (Fig. S19†). With constant potential oxidation of HMF at 1.62 V (vs. RHE), the concentration changes of HMF, intermediate products, and FDCA detected by HPLC are shown in Fig. 6a and b. The results show that the HMF concentration in the reaction medium decreased continuously with a concomitant increase of FDCA, along with the passed charges increasing during reaction. Small amounts of intermediate products, such as HMFCa and DFF, were also detected. FFCA was detected as the main accumulated intermediate product during electrolysis, indicating that the oxidation of FFCA to FDCA is the rate-determining step. The current density decreased from 70.3 mA cm<sup>-2</sup> in the beginning of reaction to 8.3 mA cm<sup>-2</sup> with the decrease of HMF concentration in the reaction medium (Fig. S20†). The passed charge was calculated during electrocatalysis and when 116 C electrons were passed in 45 min, 100% HMF conversion with 99.6% FDCA selectivity and 99.3% FE of FDCA were obtained at 1.62 V (vs. RHE). The HMF conversion and FDCA yield of the CuO-NF@Cu electrode were almost directly proportional to the passed charge (Fig. 6b and S21†).

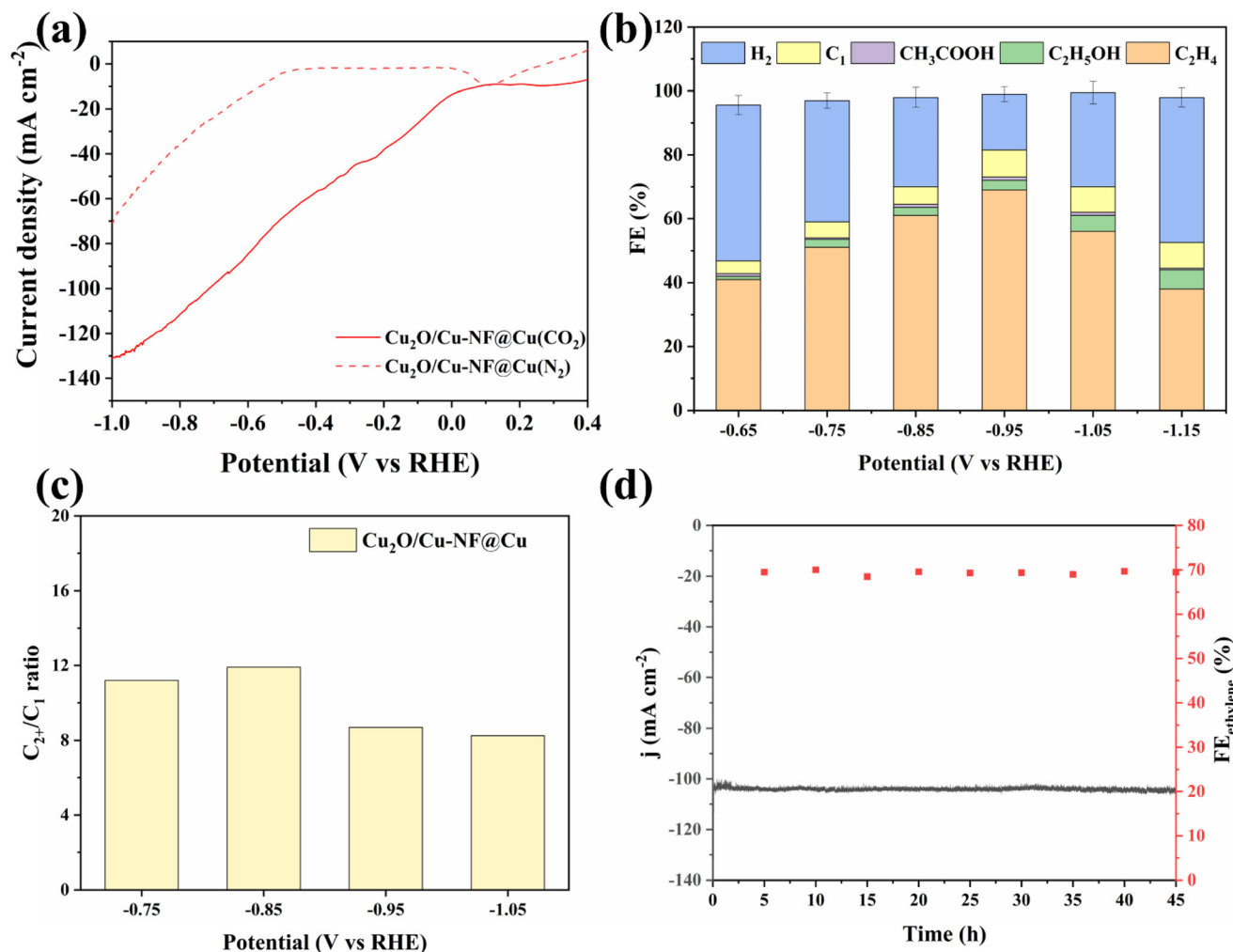


Fig. 5 (a) LSV curves over  $\text{Cu}_2\text{O}/\text{Cu-NF@Cu}$  in Ar and  $\text{CO}_2$  saturated KCl electrolyte, (b) FEs of products over  $\text{Cu}_2\text{O}/\text{Cu-NF@Cu}$ , (c)  $\text{C}_2^+/\text{C}_1$  ratios for  $\text{CO}_2\text{RR}$  products, (d) long-term stability test of 45 h at  $-0.95$  V vs. RHE for  $\text{Cu}_2\text{O}/\text{Cu-NF@Cu}$ .

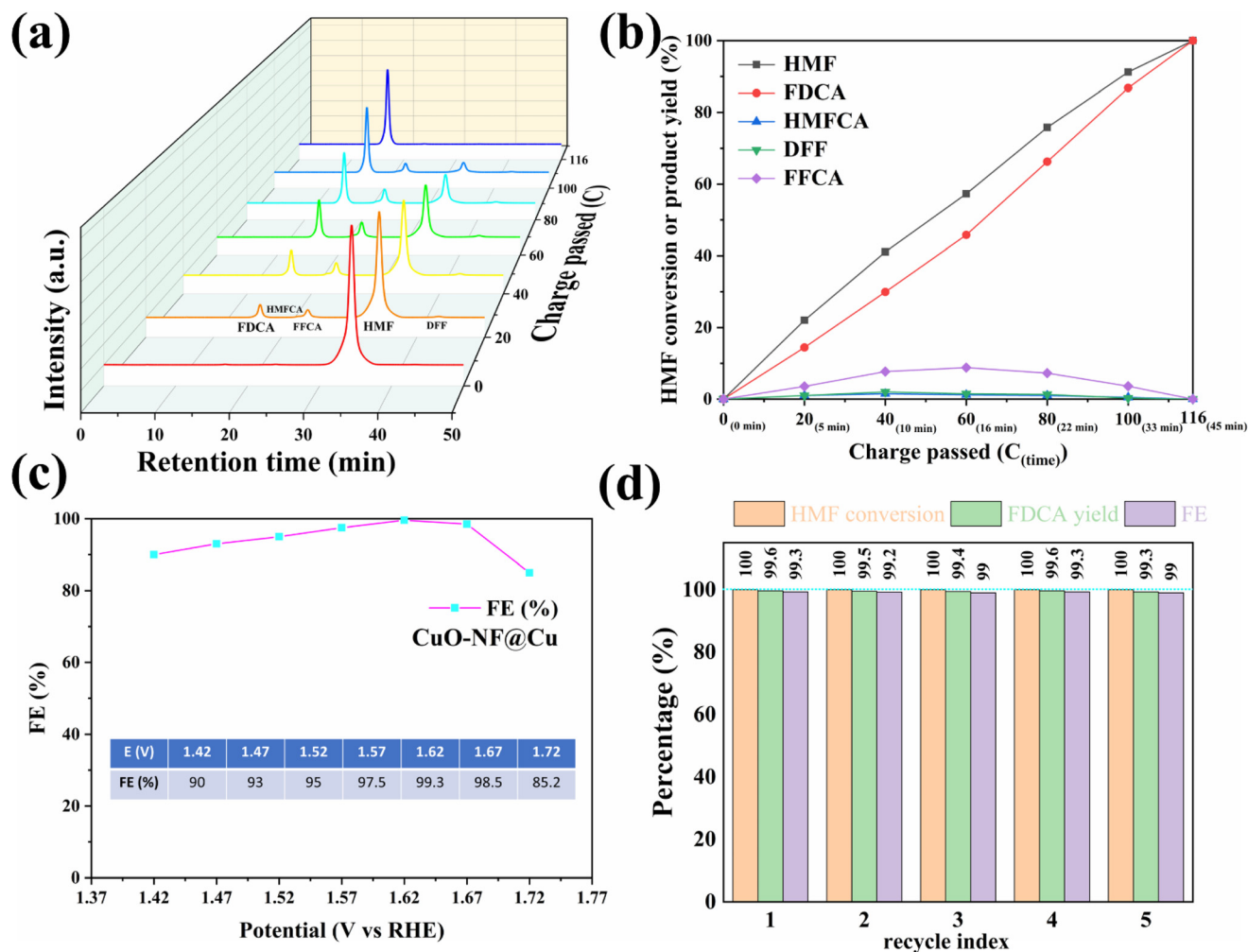
The high FE of FDCA encouraged us to check its dependence on the potential; the results in Fig. 6c show that the high FDCA FE can be maintained over a wide voltage range but started to decrease when the bias exceeded 1.67 V (vs. RHE) due to the competing OER. To study the stability of  $\text{CuO-NF@Cu}$ , five consecutive cycles of HMF oxidation were performed at a potential of 1.62 V and are shown in Fig. 6d. High and stable FDCA yield and FE were observed in the 5 recycle tests. Compared with previously reported catalysts (ESI Table S3†),  $\text{CuO-NF@Cu}$  exhibits comparable performance in the electrochemical oxidation of HMF into FDCA.

Furthermore, the crystal structure, morphology, composition, and element distribution of  $\text{CuO-NF@Cu}$  after five consecutive electro-oxidation cycles of HMF were investigated by XRD, SEM, EDS mapping, and XPS. XRD patterns (Fig. S22†) and SEM images (Fig. S23†) showed an almost identical crystal lattice and morphology of  $\text{CuO-NF@Cu}$  before and after long-term electrolysis. Moreover, the composition and element distribution of  $\text{CuO-NF@Cu}$  (Fig. S24†) and binding energy of Cu

2p (Fig. S25†) did not show significant changes after long-term electrocatalysis. All these results demonstrate the high stability of  $\text{CuO-NF@Cu}$ , which is crucial for future industrial applications.

#### Simultaneous cathodic $\text{CO}_2\text{RR}$ and anodic HMFOR over Cu-based nanoflower sheets

Encouraged by the excellent  $\text{CO}_2\text{RR}$  and HMFOR performances of the Cu-based nanoflower sheets, we pursued coupling these two half-reactions in one cell. To evaluate the potential application of the catalyst in an industrial-scale reactor, we measured their catalytic performance in a flow cell.  $\text{CuO-NF@Cu}$  was used directly as the anode electrode. In the cathode, the catalyst  $\text{CuO-NF@Cu}$  was first sprayed onto carbon paper (Sigracet 39BB) with a gas diffusion layer (GDL,  $\text{CuO-NF@GDL}$ ), followed by constant current cathodic reduction to obtain  $\text{Cu}_2\text{O}/\text{Cu-NF@GDL}$ . The detailed synthesis method is given in the ESI† ('Electrocatalysis experiments' section). The prepared  $\text{CuO-NF@GDL}$  were characterized by



**Fig. 6** (a) HPLC spectra of HMF reaction solution vs. passed charges during reaction, (b) HMF conversion and products yields vs. passed charges during reaction, (c) FE of FDCA at varied potentials, (d) recycling tests of CuO-NF@Cu.

SEM (Fig. S26 and S27†) and the results indicate that CuO-NF@GDL has same nanostructure as CuO-NF@Cu. During the flowing CO<sub>2</sub>RR-HMFOR process, the CuO-NF on GDL was reduced to Cu<sub>2</sub>O/Cu-NF (Cu<sub>2</sub>O/Cu-NF@GDL). After the reaction, the Cu<sub>2</sub>O/Cu-NF@GDL was recovered and characterized by SEM and HRTEM. Results show that Cu<sub>2</sub>O/Cu-NF@GDL has a similar nanoflower structure (Fig. S28 and S29†) to Cu<sub>2</sub>O/Cu-NF@Cu (Fig. 4e, f and Fig. S9, S10†).

The coupled CO<sub>2</sub>RR-HMFOR over Cu-based electrode was tested in a flowing setup, as shown in Fig. 7. The cathode (CuO-NF@GDL) and anode (CuO-NF@Cu) were connected to conductive copper tapes so that they could connect to an external electrochemical workstation. A bipolar membrane and the two electrodes were combined using polytetrafluoroethylene spacers so that a liquid electrolyte could be introduced into the chambers between the anode and the membrane, as well as between the membrane and the cathode. In the CO<sub>2</sub>RR compartment, gaseous CO<sub>2</sub> passed through the gas chamber at the back side of CuO-NF@GDL with a constant flow rate of 15 mL min<sup>-1</sup>. 100 mL of circular 0.5 M KCl aqueous solution

was used as the catholyte and 2 M KOH aqueous solution mixed with or without 10 mM HMF was used as the anolyte. The electrolyte flow rates at both sides were kept at 10 mL min<sup>-1</sup> during the reaction. LSV tests in Fig. 8a show that there is an obviously higher current density in coupled CO<sub>2</sub>RR-HMFOR than in CO<sub>2</sub>RR-OER (without HMF), HMFOR-HER (without CO<sub>2</sub>) or HER-OER (without CO<sub>2</sub> and HMF), indicating the overall thermodynamic advantage of the CO<sub>2</sub>RR-HMFOR process over Cu-based nanoflower sheets. Specifically, coupled CO<sub>2</sub>RR-HMFOR provided a current density of 100 mA cm<sup>-2</sup> at a cell voltage of 2.31 V, which was 790 mV lower than the voltage of CO<sub>2</sub>RR-OER (3.1 V). At cell voltages of 2.5, 2.75, and 3 V, the timed-current curves showed high current densities of 134.5, 188.8 and 245.1 mA cm<sup>-2</sup> (Fig. 8b), respectively. The corresponding FE<sub>C<sub>2</sub>H<sub>4</sub></sub>/FE<sub>FDCA</sub> values in coupled CO<sub>2</sub>RR-HMFOR were calculated to be 55.6%/85.5%, 74.5%/96.6%, and 58.6%/88.3%, respectively (Fig. 8c). The stability test of the Cu-based nanoflower sheet catalysts in coupled CO<sub>2</sub>RR-HMFOR gave a highly stable current density of about 190.0 ± 5.2 mA cm<sup>-2</sup> within 5 hours and the FE<sub>C<sub>2</sub>H<sub>4</sub></sub>/



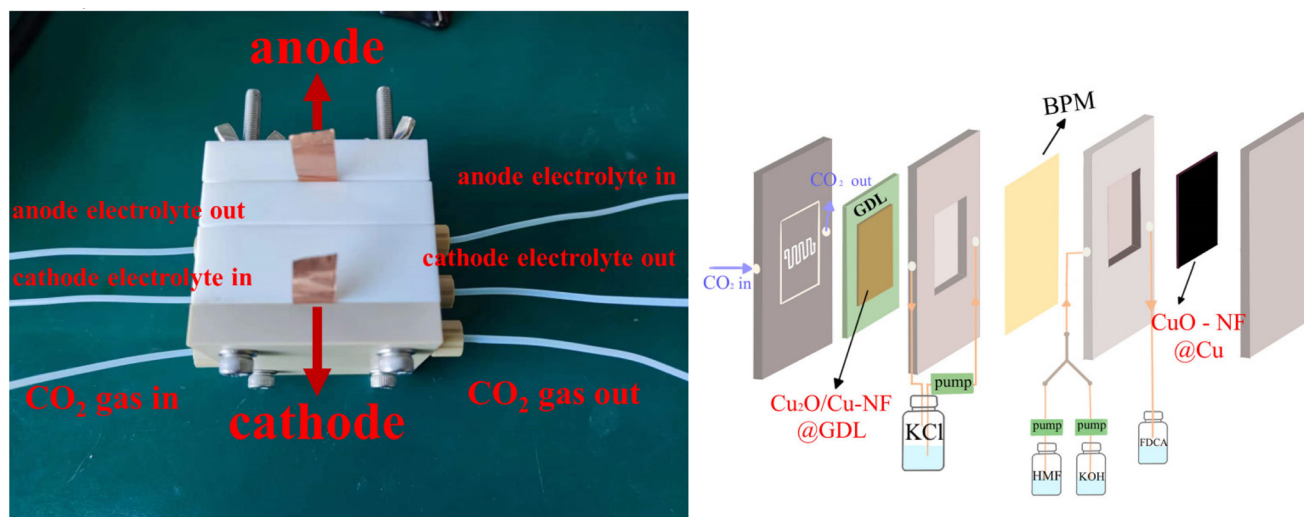


Fig. 7 Photo and schematic illustration of the used flow cell.

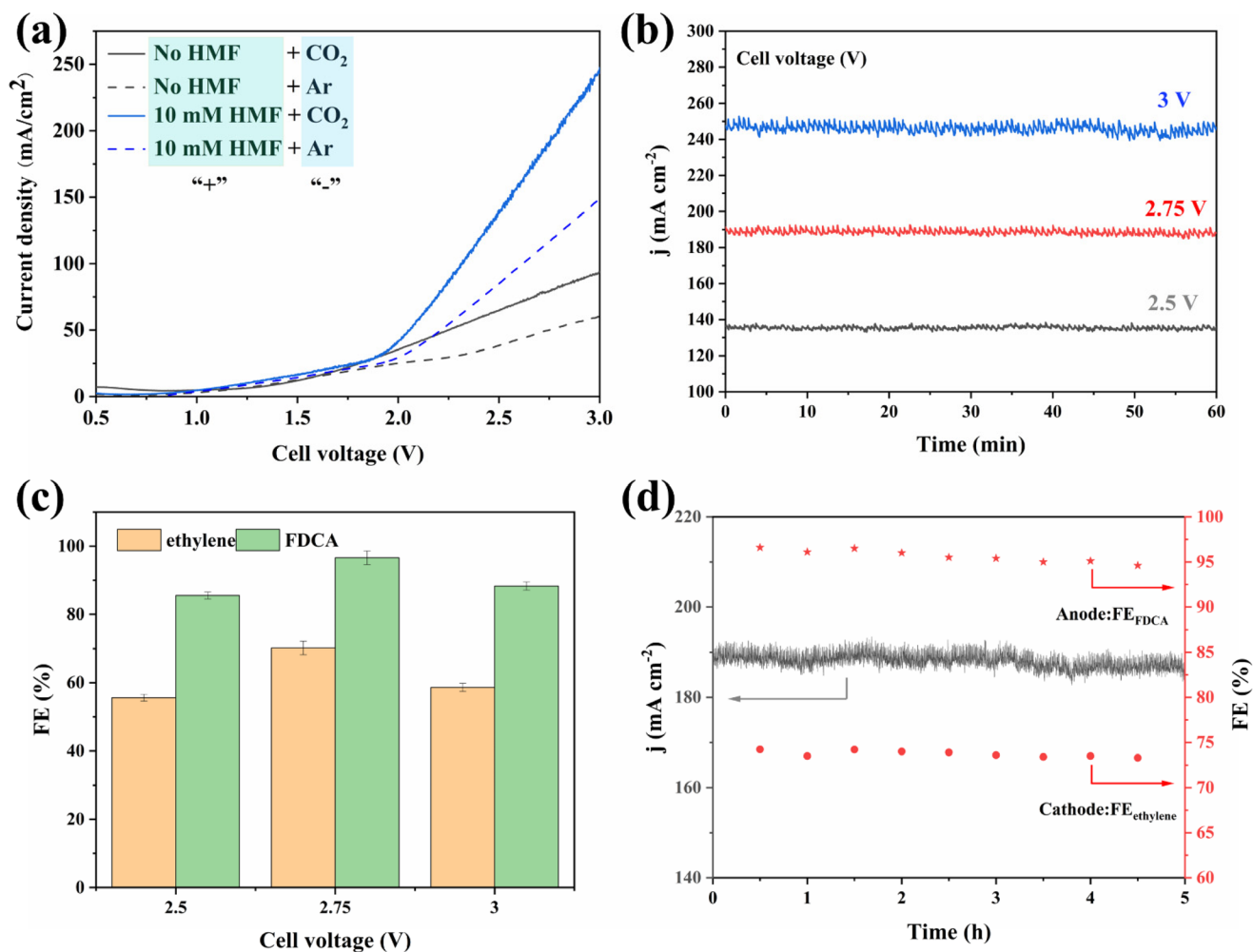


Fig. 8 (a) LSV curves recorded in two-electrode setup. "+": anodic compartment; "-": cathodic compartment. (b) Chronoamperometry curves under different cell voltages, (c) corresponding FE for CO<sub>2</sub>RR-HMFOR couple under different cell voltages, (d) stability test of Cu-based nanoflower sheets in CO<sub>2</sub>RR-HMFOR coupled reaction under a cell voltage of 2.75 V.

FE<sub>FDCA</sub> values were maintained at *ca.* 71 ± 3.8%/93 ± 3.6% (Fig. 8d). The anode CuO-NF@Cu in the coupled reaction was recovered after 5 h and characterized by SEM; the results (Fig. S30†) show that the recovered CuO-NF@Cu maintained its original nanostructure, as shown in Fig. 2 and 3. Thus, the high FE of FDCA and C<sub>2</sub>H<sub>4</sub> in coupled CO<sub>2</sub>RR-HMFOR can be attributed to the stable nanostructure of the CuO nanoflower sheets in CuO-NF@Cu and the stable Cu/Cu<sup>+</sup> interface in Cu<sub>2</sub>O/Cu-NF@Cu (Fig. S9 and S10†).

## Conclusions

In this work, foam Cu was easily modified to a nanoflower CuO-covered Cu foam catalyst, where CuO is composed of a stable structure of nanosheets. In CO<sub>2</sub>RR, CuO-NF@Cu can be quickly reduced to a nanoflower-shaped Cu<sub>2</sub>O/Cu-NF@Cu, which has a stable nanoparticle structure and stable Cu<sup>+</sup>/Cu interface. At a reduction potential of −0.95 V (*vs.* RHE), a high C<sub>2</sub>H<sub>4</sub> FE up to 70% with a current density above 100 mA cm<sup>−2</sup> can be obtained. In HMFOR, CuO-NF@Cu shows excellent catalytic activity in the oxidation of HMF into FDCA. At a constant potential of 1.62 V (*vs.* RHE) and passed charge of 116 C, the FE of FDCA reached 99.3%. In coupled CO<sub>2</sub>RR-HMFOR with the above Cu-based nanoflower electrocatalysts, a current density up to 188.8 mA cm<sup>−2</sup> at a cell voltage of 2.75 V can be obtained. Moreover, FE values of FDCA and C<sub>2</sub>H<sub>4</sub> up to 74.5% and 96.6%, respectively, in the coupled CO<sub>2</sub>RR-HMFOR were achieved. The microstructure of the Cu-based nanoflower catalyst maintained stability and gave above 180.0 mA cm<sup>−2</sup> current density within 5 h of testing. This work shows that an easily prepared Cu-based catalyst can be used as a highly efficient electrocatalyst to achieve high-value products by the oxidation of biomass platform compounds in the anode and the reduction of CO<sub>2</sub> into high-value C<sub>2</sub>H<sub>4</sub> in the cathode. With the development of efficient electrocatalysts, renewable energy can be used to achieve the resource utilization of CO<sub>2</sub> produced from fossil resources while also upgrading renewable biomass resources.

## Conflicts of interest

There are no conflicts to declare.

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