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# High-efficiency electrodeposition of magnesium alloy-based anodes for ultra-stable rechargeable magnesium-ion batteries†

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Rechargeable magnesium batteries (RMBs) have attracted much attention because of their high theoretical volumetric capacity and high safety. However, the uneven deposition behavior, harmful corrosion reaction and poor stability of magnesium metal anodes have hindered the practical application of RMBs. Herein, we propose a facile alloy electrodeposition method to construct an artificial layer on an Mg anode. Experimental results show that the polarization of the symmetric magnesium alloy-based (Mg–Sn@Mg and Mg–Bi@Mg) cells is significantly reduced (~0.05 V) at a current density of 0.1 mA cm$^{-2}$. The symmetric cells using the prepared Mg alloy anodes exhibited lower voltage hysteresis and ultra-stable cycling performance at a higher density of 1.0 mA cm$^{-2}$ over 700 h. The *in situ* optical microscopy study clearly demonstrated that the Mg dendrite formation was successfully retarded by the designed Mg–Sn and Mg–Bi alloy artificial protective layer on Mg anodes. The superiority of Mg–Sn@Mg and Mg–Bi@Mg was further confirmed in full cells using $Mo_6S_8$ as the cathode. Compared with the $Mo_6S_8$//Mg full cell, the $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg full cells maintained an ultra-stable electrochemical performance even after 5000 cycles. This proof-of-concept provides a novel scope for the artificial coating layers on Mg anodes prepared by alloy electrodeposition and can be extended to other alloy anodes (*i.e.* Mg–Cu@Mg and so on). This work provides an avenue for the design of practical and high-performance RMBs and beyond.

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## Introduction

Growing concerns about global environmental and energy sustainability are now driving research and development of clean energy storage technologies.[1–4] The dominant Li-ion batteries (LIBs) in the energy storage landscape are facing constraints in meeting the development of vehicle electrification and large-scale energy storage due to their limited energy density and scarce lithium resources.[5–9] Rechargeable magnesium-ion batteries (RMBs) have attracted increasing attention due to their high capacity, natural abundance, low cost and environmental friendliness to meet the rapid demand for renewable energy storage systems.[10–13]

Magnesium metal has a low redox potential (−2.37 V *vs.* SHE), a high volumetric capacity (3733 mA h cm$^{-3}$), good chemical stability and high terrestrial abundance.[9,14–16] Aurbach *et al.* created the initial magnesium battery prototype in 2000.[17] Although there has been significant advancement in research on cathodes and electrolytes since then, one of the main obstacles to magnesium ion battery development remains to be the incompatibility of anodes, electrolytes, and cathodes.[18–20] However, unlike other metal anodes (such as lithium), the usage of magnesium metal anodes presents a far greater challenge due to the creation of a passivation layer on magnesium metal in the majority of organic electrolytes.[21–23] This $Mg^{2+}$ passivation layer, which consists mainly of magnesium oxide (MgO) or magnesium hydroxide ($Mg(OH)_2$), is thought to form naturally on the magnesium surface and can lead to irreversible magnesium plating/stripping.[12,24,25] Therefore, a great deal of research has focused on the invention of new electrolytes to reduce the formation of insulating layers in the early stages of research.[24,26–28] Facing passivation problems in conventional $Mg(TFSI)_2$-based electrolytes and the issue of "magnesium dendrites" in Grignard's agents or an all-phenyl compound (APC) under extreme conditions (high current density or large areal capacity), it is crucial to develop magnesium-metal anodes that are not passivated and free of dendrites in conventional electrolytes.[8,29–31]

The current optimization strategies for Mg anodes mainly include electrolyte modification,[32–34] constructing the metal–

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organic skeleton,[35,36] and designing an artificial solid electrolyte interphase (SEI).[37–39] In particular, in recent years, some researchers have modified magnesium anodes by designing artificial SEIs, which can enhance the stability of the anode interface. Li *et al.*[40] added $I_2$ to the electrolyte to form a $MgI_2$-based protective layer on the magnesium anode. Zhang *et al.*[41] developed a Ge-based alloy layer on the Mg metal anode by adding an excessive amount of $GeCl_4$. Li *et al.*[42] designed a protective layer on a Mg metal anode by applying $SiCl_4$/DME solution, and a protective layer enriched with $MgCl_2$ at the top and Si at the bottom was designed. Wang *et al.*[43] electrochemically pretreated magnesium and a DOL-based SEI was formed on the Mg metal surface. These efforts can effectively inhibit the surface passivation between Mg and the electrolyte and improve the kinetics of Mg plating/stripping, thus improving cycling stability and multiplicative performance.[44,45] However, the majority of magnesium anode modification methods usually require complex preparation processes and pretreatment procedures and are expensive to prepare.[46,47] Electrodeposition is a centuries-old technique for the synthesis of inorganic materials, conductive polymers and metal–organics.[48–52] It is a simple, low-cost, efficient and easy-to-control method for the preparation of surface layers.[52,53] Simultaneously, electrodeposition is a simple, low-cost, efficient and easy-to-control technique for producing thin metal films on conductive substrates.[54]

In this work, a facile and high-efficiency electrodeposition was used to deposit Mg–Sn@Mg and Mg–Bi@Mg artificial protective layers on Mg metal surfaces, which can be stably cycled for a long time in an APC electrolyte. The Mg–Sn@Mg and Mg–Bi@Mg layers offer a rapid channel for the transport of $Mg^{2+}$, and they can also significantly enhance the inhomogeneous Mg plating/stripping behavior, hence facilitating the uniform deposition of $Mg^{2+}$ and enhancing the performance of RMBs. Hence, Mg–Sn@Mg and Mg–Bi@Mg demonstrate a remarkable long life of over 1000 h with a low polarization voltage (0.2 mA $cm^{-2}$; 0.2 mA h $cm^{-2}$) and excellent multiplicative performance (at different current densities from 0.1 to 1 mA $cm^{-2}$) in symmetric cell tests. Even at high current densities (1.0 mA $cm^{-2}$; 1.0 mA h $cm^{-2}$), the Mg–Bi@Mg and Mg–Sn@Mg symmetric cells exhibit enhanced cyclability and lower polarization voltages than the bare Mg symmetric cell. In addition, when coupled with $Mo_6S_8$ cathodes, the full cells exhibit extraordinary electrochemical performance even after 5000 cycles at 10 C.

## Results and discussion

The Mg–Sn@Mg and Mg–Bi@Mg anodes were prepared using alloy electrodeposition (detailed information is provided in the Experimental section). As shown in the Scanning Electron Microscopy (SEM) images of Fig. 1a and b, homogeneous coatings of Mg–Sn@Mg and Mg–Bi@Mg with a 3D structure were formed on Mg surfaces after alloy electrodeposition. The EDS elemental mapping images (Fig. 1c) show the uniform distribution of the two elements (Mg, Sn and Mg, Bi) on the surface of magnesium metal, respectively. The cross-sectional SEM images (Fig. S1 and S2†) show that the Mg alloy layers are densely and uniformly adhered to the Mg substrate. The thicknesses of Mg–Sn and Mg–Bi are 3.5 μm and 2.2 μm, respectively. The compositions of Mg–Sn@Mg and Mg–Bi@Mg were identified by X-ray diffraction (XRD) measurements. Fig. 1d and g confirm that Mg, Sn, and Mg, Bi metals are observed in the corresponding protective layers, respectively (Mg (JCPDS no. 35-0821), Bi (JCPDS no. 44-1246) and Sn (JCPDS no. 65-0296)). The above observations suggest that Mg–Sn@Mg and Mg–Bi@Mg anodes were successfully prepared by this simple alloy electrodeposition. We also used alloy electrodeposition to prepare different protective layers on Mg foil, such as Mg–Cu@Mg. XRD pattern demonstrates the Mg–Cu@Mg coating as shown in Fig. S3.† The SEM image in Fig. S4† shows that the Mg–Cu protective layer is uniformly distributed on the magnesium substrate.

X-ray photoelectron spectroscopy (XPS) was used to characterize Mg–Sn@Mg and Mg–Bi@Mg anodes. The full XPS spectrum of Mg–Sn@Mg in Fig. S5† indicates the presence of the elements Mg and Sn in the as-prepared sample after alloy electrodeposition. The high-resolution XPS spectrum of the Mg 1s region (Fig. 1e and h) can be fitted to a Gaussian peak centered at 1303.5 eV, indicating the presence of Mg. The XPS spectrum of the Sn 3d region (Fig. 1f) can be resolved into two peaks, Sn $3d_{5/2}$ and Sn $3d_{3/2}$, with high resolution corresponding to binding energies of 486.5 eV and 495 eV, respectively. Similarly, the full XPS spectrum of Mg–Bi@Mg in Fig. S6† shows the presence of Mg and Bi elements in the sample after alloy electrodeposition. The high-resolution XPS spectrum of the Bi 4f region (Fig. 1i) can be resolved into four peaks corresponding to four Gaussian peaks centered at 156.0 eV, 158.5 eV, 161.2 eV, and 163.8 eV, respectively. Corresponding to the XPS results of bare Mg in Fig. S7,† it reconfirms the successful preparation of Mg–Sn@Mg and Mg–Bi@Mg anodes.

The all-phenyl complex (APC) electrolyte was used in RMBs. The plating/stripping process of the APC electrolyte was tested using a three-electrode cell (Fig. S8†). The reversibility of the magnesium plating/stripping reaction was tested using a symmetric cell configuration consisting of two identical magnesium electrodes in an APC electrolyte. Fig. 2a shows the voltage profiles of the Mg–Sn@Mg//Mg–Sn@Mg, Mg–Bi@Mg//Mg–Bi@Mg and bare Mg//bare Mg symmetric cells at a current density of 0.1 mA $cm^{-2}$ (0.1 mA h $cm^{-2}$). The bare Mg electrode showed an extremely high initial polarization of 0.36 V (Fig. 2b). The polarization gradually decreased during further cycles, reaching ~0.10 V at the 25th cycle. In contrast, both Mg–Sn@Mg and Mg–Bi@Mg electrodes achieved high operational stability of more than 1000 h with slight voltage fluctuations, and stabilized at ~0.05 V. This indicates that due to the presence of the alloy protective layer, the electrochemical activity and reversibility of the Mg metal anode are improved. In addition, Mg–Cu@Mg symmetric cells were assembled to verify that the Mg plating/stripping was reversible. As shown in

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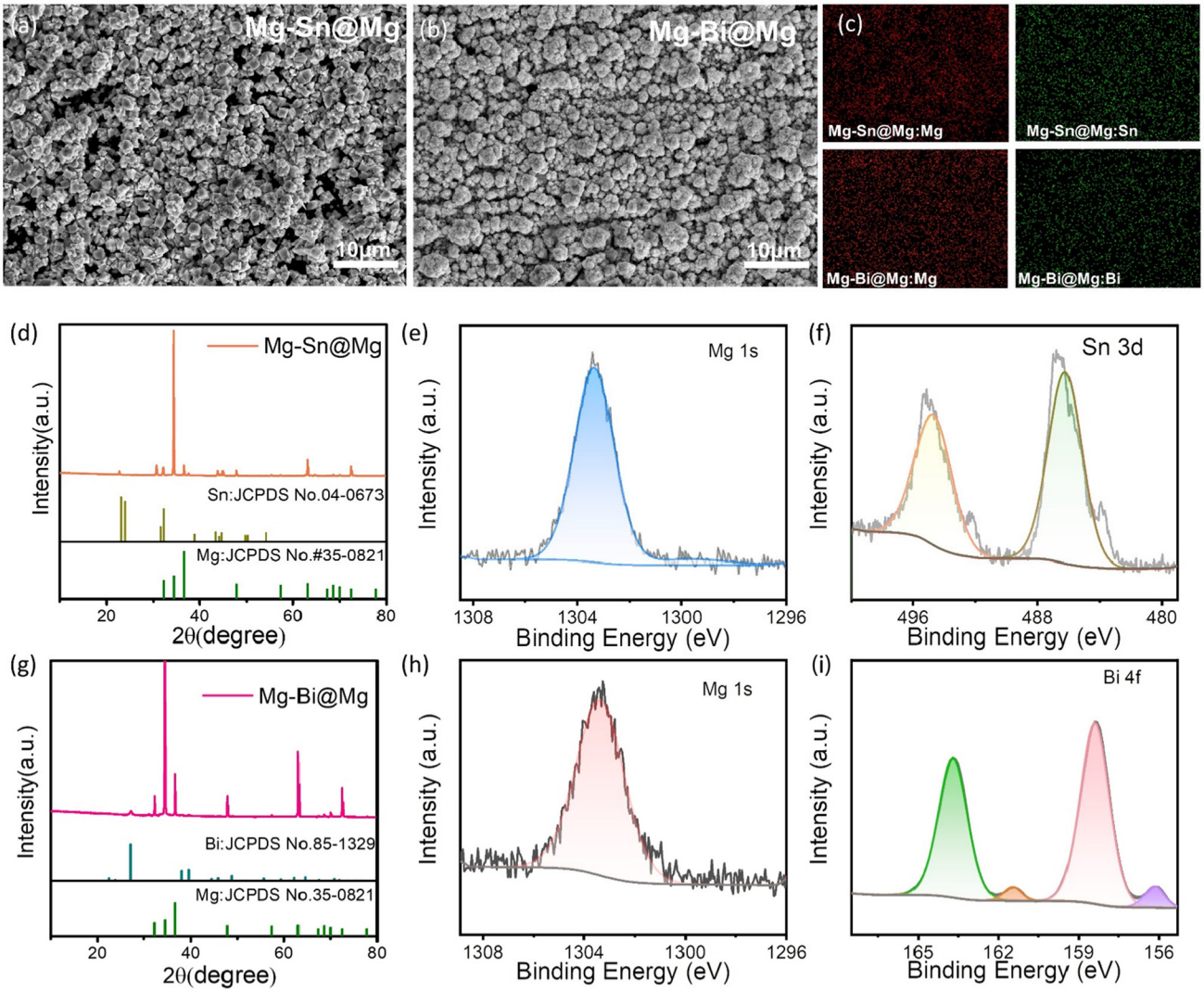


**Fig. 1** SEM images of (a) Mg–Sn@Mg and (b) Mg–Bi@Mg. (c) EDS elemental mapping images of Mg–Sn@Mg and Mg–Bi@Mg. (d) XRD patterns of Mg–Sn@Mg. XPS analysis of the survey spectrum: (e) Mg 1s and (f) Sn 3d. (g) XRD patterns of Mg–Bi @Mg. XPS analysis of the survey spectrum: (h) Mg 1s and (i) Bi 4f.

Fig. S9,† at a current density of 0.1 mA $cm^{-2}$, Mg–Cu@Mg//Mg–Cu@Mg achieves a stable cycle of more than 700 h and a low polarization of 0.065 V.

Fig. 2c–f show the reversibility of Mg plating/stripping behaviors at different current densities, respectively. When the areal capacity was increased to 0.2 mA h $cm^{-2}$ (Fig. 2c), the voltage hysteresis of the bare magnesium cell showed serious fluctuations during cycling, and the voltage suddenly plummeted after 619 h. A sharp drop in voltage in a symmetrical cell is usually considered a sign of a short circuit.[55–57] Similarly, the polarization of the Mg–Sn@Mg and Mg–Bi@Mg symmetric cells gradually decreased and was lower than that of bare Mg cells. The Mg–Sn@Mg and Mg–Bi@Mg anodes had a low voltage hysteresis of ≈0.07 V (Fig. 2d) and negligible voltage fluctuations for more than 1000 h. The superiority of Mg–Sn@Mg and Mg–Bi@Mg in particular was again demonstrated at a high current density of 1 mA $cm^{-2}$ and a higher areal capacity of 1 mA h $cm^{-2}$. As shown in Fig. 2e and f, the bare Mg symmetric cell suffers from severe voltage fluctuations and only lasts for 125 h before short-circuiting. Impressively, the Mg–Sn@Mg and Mg–Bi@Mg cells were able to achieve stability at low average overpotentials of 0.18 V and 0.14 V, respectively, for more than 700 h. The electrochemical performance of the above symmetric cells demonstrates that bare Mg has higher voltage hysteresis and shorter lifetime compared to Mg–Sn@Mg and Mg–Bi@Mg. It is verified that the electrochemical activity and reversibility of magnesium-based alloy anodes (Mg–Sn@Mg and Mg–Bi@Mg) are substantially improved compared with those of bare Mg metal anodes.

Galvanostatic tests on symmetric cells evaluated the superiority of Mg–Sn@Mg and Mg–Bi@Mg in terms of cycling stability. The rate capability of the symmetric cells (bare Mg//bare

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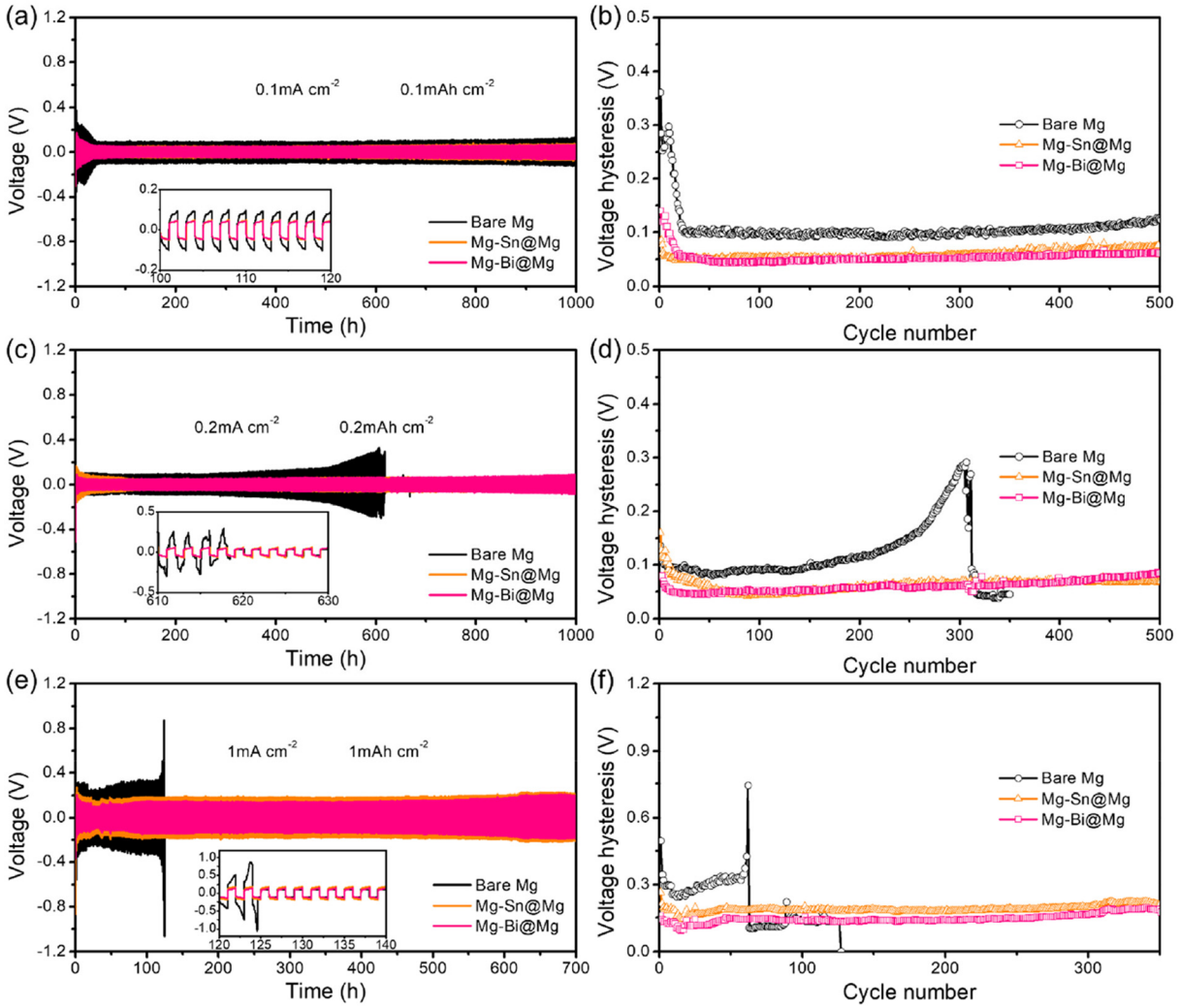


**Fig. 2** (a) Cycling performance and (b) the corresponding voltage hysteresis of symmetric cells with bare Mg, Mg–Sn@Mg and Mg–Bi@Mg electrodes at 0.1 mA cm$^{-2}$ (areal capacity: 0.1 mA h cm$^{-2}$). (c) Cycling performance and (d) the corresponding voltage hysteresis of symmetric cells with bare Mg, Mg–Sn@Mg and Mg–Bi@Mg electrodes at 0.2 mA cm$^{-2}$ (0.2 mA h cm$^{-2}$). (e) Cycling performance and (f) the corresponding voltage hysteresis of symmetric cells with bare Mg, Mg–Sn@Mg and Mg–Bi@Mg electrodes at 1 mA cm$^{-2}$ (areal capacity: 1.0 mA h cm$^{-2}$).

Mg, Mg–Sn@Mg//Mg–Sn@Mg, Mg–Bi@Mg//Mg–Bi@Mg) was verified at different current densities from 0.1 to 1.0 mA cm$^{-2}$. The polarization voltages of the cells with Mg–Sn@Mg and Mg–Bi@Mg are consistently lower than that of the cells with bare Mg at all current densities (Fig. 3a). More importantly, the higher the applied current density, the greater the difference in polarization voltages between the three cells. Furthermore, the magnesium deposition kinetics can be quantified by the exchange current density ($i_0$), which is calculated according to the following equation:[58]

$$i = i_0 \frac{F}{RT} \cdot \frac{\eta}{2} \tag{1}$$

where $i$ is the operating current density, $F$ is Faraday's constant, $\eta$ is the total overpotential, $R$ is the gas constant, and $T$ is the absolute temperature.

The total overpotential was obtained from the electrostatic tests of symmetrical cells at different current densities (Fig. 3a). As shown in Fig. 3b, the $i_0$ of Mg–Sn@Mg and Mg–Bi@Mg is 0.211 and 0.243 mA cm$^{-2}$, respectively, which is much higher than that of bare Mg (0.136 mA cm$^{-2}$). The kinetics of Mg deposition enhanced by Mg–Sn@Mg and Mg–Bi@Mg shows that Mg–Sn@Mg and Mg–Bi@Mg can effectively improve the Mg deposition/dissolution behavior. To study the electrochemical dynamics and performance of Mg–Sn@Mg and Mg–Bi@Mg with stable Mg deposition/dissolution behavior, various electrochemical analyses were carried out on symmetric cells. Cyclic voltammetry (CV) measurements of the symmetric cells were performed at a scan rate of 1 mV s$^{-1}$. Fig. 3c shows that the combined peak areas of Mg–Sn@Mg and Mg–Bi@Mg symmetric cells are larger than that of bare Mg cell, which indicates the improved electrochemical redox activity of Mg deposition. From the contact angle tests, the

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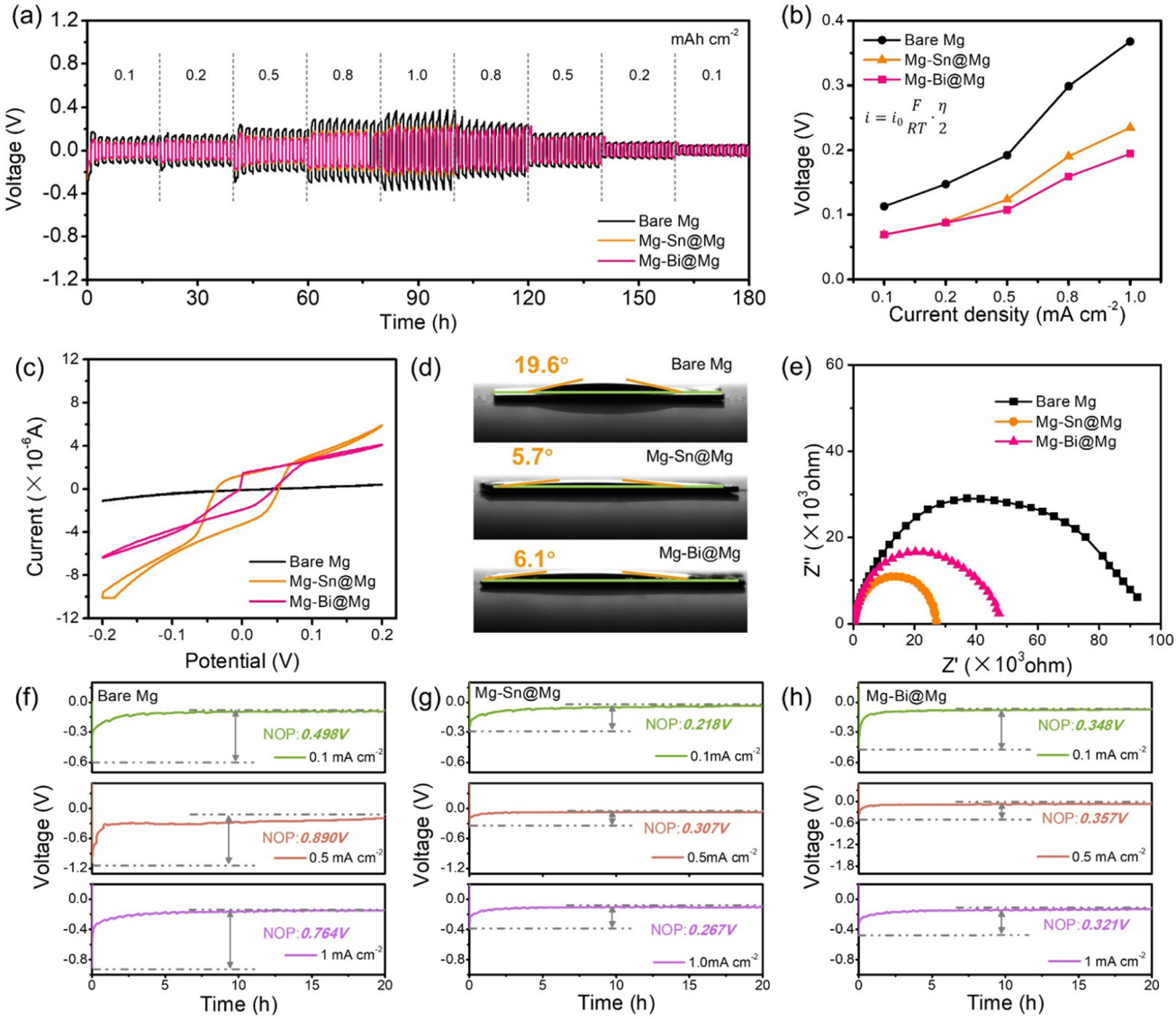


**Fig. 3** (a) Rate capability of symmetric cells at different current densities from 0.1 to 2.0 mA $cm^{-2}$ and (b) corresponding exchange current densities of bare Mg, Mg–Sn@Mg and Mg–Bi@Mg. (c) CV curves of symmetric cells (bare Mg//bare Mg, Mg–Sn@Mg//Mg–Sn@Mg and Mg–Bi@Mg//Mg–Bi@Mg) obtained at a scan rate of 1 mV $s^{-1}$. (d) Contact angles of the APC electrolyte on bare Mg, Mg–Sn@Mg and Mg–Bi@Mg, respectively. (e) EIS for the Mg–Mg symmetric cells using bare Mg, Mg–Sn@Mg and Mg–Bi@Mg. (f–h) Voltage profiles of Mg deposition on (f) bare Mg, (g) Mg–Sn@Mg and (h) Mg–Bi@Mg at different current densities from 0.1 to 1 mA $cm^{-2}$.

interfacial performance of the Mg anode was examined by whether the anode surface sufficiently absorbed the APC electrolyte. As shown in Fig. 3d, the affinity performance of Mg–Sn@Mg (5.7°) and Mg–Bi@Mg (6.1°) was enhanced compared to that of bare Mg (19.6°). Similarly, the affinity property of Mg–Cu@Mg (4.1°) is also enhanced as shown in Fig. S10.† This suggests that the alloy-based protective layer prepared by alloy electrodeposition may provide abundant active sites and enhance the transport/diffusion of $Mg^{2+}$ ions. To further confirm the above findings, electrochemical impedance spectroscopy (EIS) analysis of the symmetric cells was performed to verify the enhanced electrochemical stability and deposition kinetics of Mg–Sn@Mg and Mg–Bi@Mg (Fig. 3e and Table S1†). Compared to the charge transfer resistance ($R_{ct}$) of the bare Mg symmetric cell (85 067 Ω), the $R_{ct}$ of the symmetric cells of Mg–Sn@Mg (26 000 Ω) and Mg–Bi@Mg (40 686 Ω) is significantly reduced. This situation suggests that Mg–Sn@Mg and Mg–Bi@Mg can promote the charge transfer kinetics and guide the deposition of Mg at the interface between the host and the electrolyte.

In addition to improving the kinetics of Mg plating/stripping, Mg–Sn@Mg and Mg–Bi@Mg also play a crucial role in reducing the Mg nucleation overpotential (NOP: the difference between the voltage dip and the stable voltage plateau). As shown in Fig. 3f and g, the NOP of Mg–Sn@Mg at a current density of 0.1 mA $cm^{-2}$ has a very low value of 0.218 V com-

pared to that of bare Mg (0.498 V). Similarly, the NOP of Mg–Bi@Mg also has a lower value of 0.348 V at 0.1 mA $cm^{-2}$ (Fig. 3h). The NOP values of Mg–Sn@Mg and Mg–Bi@Mg exhibit less fluctuation as the current density increases. Even at a high current density of 1.0 mA $cm^{-2}$, Mg–Sn@Mg and Mg–Bi@Mg showed lower NOP values (Fig. S11†). In contrast, the NOP of bare Mg rapidly increased from 0.498 V to 0.764 V with the increase of the current density, showing a great difference in NOP values among Mg–Sn@Mg, Mg–Bi@Mg and bare Mg. According to these findings, Mg–Sn@Mg and Mg–Bi@Mg provide abundant nucleation sites as cell electrodes, which ultimately ensures more uniform Mg deposition.

EIS was used to further investigate the interfacial resistance of the symmetric cells in different cycles. The $R_{ct}$ values of Mg–Sn@Mg and Mg–Bi@Mg (Fig. 4a–c and Tables S2–4†) in the initial state are 16 171 Ω and 58 125 Ω, respectively, which are much lower than that of the bare Mg anode (85 067 Ω). After 20 cycles, the interfacial adsorption layer of bare Mg symmetric cell may be gradually detached or destroyed due to Mg deposition, which exhibits lower interfacial impedance during the

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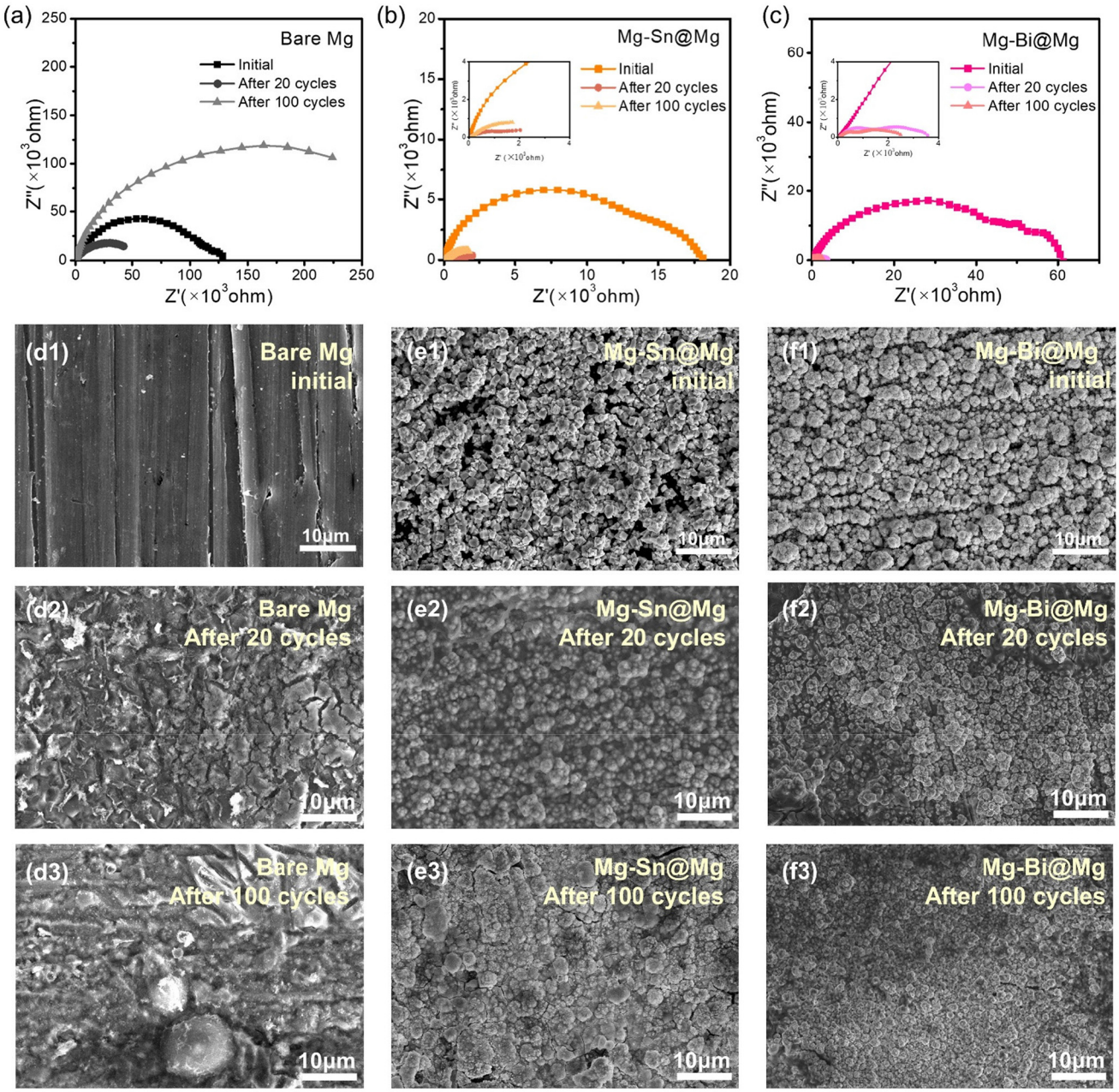


**Fig. 4** (a–c) EIS for the Mg//Mg symmetric cells using (a) bare Mg, (b) Mg–Sn@Mg and (c) Mg–Bi@Mg after various cycles. SEM image of (d1–d3) bare Mg, (e1–e3) Mg–Sn@Mg and (f1–f3) Mg–Bi@Mg electrodes in symmetric cells after different cycles.

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cell deposition process.[38] As shown in Fig. 4a–c, the charge transfer resistance of bare Mg decreases after 20 cycles. The $R_{ct}$ values of both Mg–Sn@Mg (1696 Ω) and Mg–Bi@Mg (2782 Ω) are much lower than that of bare Mg cells (60 077 Ω). However, the $R_{ct}$ of bare Mg (189 780 Ω) increases substantially after 100 cycles. The impedance of Mg–Sn@Mg (4559 Ω) and Mg–Bi@Mg (2184 Ω) still maintains a low $R_{ct}$ even after 100 cycles, which suggests that Mg–Sn and Mg–Bi alloy layers can stabilize the interface of the anode/electrolyte. Furthermore, the evolution of the morphology of plated magnesium electrodes with an areal capacity of 0.1 mA h $cm^{-2}$ was examined with various cycles using SEM (Fig. 4d1–d3, e1–e3, and f1–f3), and the transformation process of the produced protective layers of Mg–Sn@Mg and Mg–Bi@Mg was visible. Pits started to form on the magnesium metal anode after the 20 plating/stripping cycles, and they tended to get bigger and enlarge throughout the following cycles (Fig. 4d3). These pits may disrupt the integrity of the Mg anode, leading to the formation of an inhomogeneous Mg coating during subsequent cycling.[38] Interestingly, compared to bare Mg, the surfaces of Mg–Sn@Mg and Mg–Bi@Mg are relatively smooth and still mainly retain their initial structures (Fig. 4e2, e3 and f2, f3). After 20 and 100 cycles, the structure of the Mg–Sn@Mg and Mg–Bi@Mg anodes remained relatively compact and consistent. Even after 200 cycles, the Mg–Sn@Mg and Mg–Bi@Mg anodes presented a flat and homogeneous surface (Fig. S12 and S13†). This morphological analysis further indicates that Mg deposition is more favorable in the structures of the Mg–Sn and Mg–Bi alloy-based protective layers for Mg–Sn@Mg and Mg–Bi@Mg anodes.

Fig. S14 and S15† show the XRD patterns of Mg–Sn@Mg and Mg–Bi@Mg after magnesiation and 10 cycles. At the magnesiated state, $Mg_2Sn$ (JCPDS no. 07-0274) can be found in the XRD pattern of the magnesiated Mg–Sn@Mg anode (Fig. S14†), and $Mg_3Bi_2$ (JCPDS no. 04-0464) is found in the XRD pattern of the magnesiated Mg–Bi@Mg anode (Fig. S15†). After 10 cycles, $Mg_2Sn$ and $Mg_3Bi_2$ are still observed in the Mg–Sn@Mg and Mg–Bi@Mg anodes, respectively. Therefore, $Mg^{2+}$ maybe undergoes alloying reactions with Sn and Bi during cycling, respectively. The reaction equations are as follows:

$$3Mg^{2+} + 2Bi + 6e^- \rightleftharpoons Mg_3Bi_2 \quad (2)$$

$$2Mg^{2+} + Sn + 4e^- \rightleftharpoons Mg_2Sn \quad (3)$$

To further characterize the deposition morphology on the Mg anode surface, the different behaviors of Mg-X alloy-based and bare Mg anodes during magnesium electroplating were visualized by *in situ* optical microscopy using a symmetric cell (ESI Movie 1†). Fig. 5a, b and c show the Mg deposition process for Mg–Sn@Mg, Mg–Bi@Mg and bare Mg. Obviously, metal Mg was densely and uniformly deposited on the anode surfaces of Mg–Sn@Mg and Mg–Bi@Mg without dendrite growth within 30 min at a current density of 4.0 mA $cm^{-2}$. It is also demonstrated that the magnesium deposition behavior is predicted to be regulated by the Mg–Sn@Mg and Mg–Bi@Mg alloy anodes. In contrast, bare Mg showed uneven Mg deposits only after 5 min of Mg plating. As plating time increases, these protrusions rapidly grow into dendrites due to uneven electric field distribution and increasing tip effect (Fig. 5c).

SEM images, XRD patterns and *in situ* optical visualization observations mentioned above demonstrate that the deposits are uniformly distributed on the surface of the Mg–Sn@Mg and Mg–Bi@Mg electrodes without large protrusions during the plating process (Fig. 5d). In contrast, bare Mg electrode produced greatly inhomogeneous deposits and the surface was severely damaged after several cycles. In addition, the presence of inhomogeneous Mg deposits leads to a porous "dead" Mg layer during cycling, which can lead to short-circuiting of the battery and other safety hazards.[59]

To further evaluate the practical applicability of Mg–Sn@Mg and Mg–Bi@Mg, we prepared and tested the full cells using $Mo_6S_8$ as the cathode. The full cells were measured in an APC electrolyte. First, the full cells were evaluated using cyclic voltammetry (CV). The initial CV curves (Fig. S16†) of $Mo_6S_8$//bare Mg, $Mo_6S_8$//Mg–Sn@Mg, and $Mo_6S_8$//Mg–Bi@Mg were tested at 0.2 mV $s^{-1}$ in the voltage range of 0.2–1.95 V. Both $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg exhibit a decrease in voltage polarization (≈0.620 V and 0.473 V, respectively). This suggests that the full-cell reaction kinetics of Mg–Sn@Mg and Mg–Bi@Mg and electrochemical performance were significantly improved. We also performed CV tests at different scan rates from 0.1 to 1.0 mV $s^{-1}$ (Fig. 6a–c) to calculate the $Mg^{2+}$ diffusion coefficients ($D_{Mg^{2+}}$) based on the Randles–Sevcik equation:[60]

$$I_{peak} = 2.69 \times 10^5 n^{1.5} A D_{Mg^{2+}}{}^{0.5} C_{Mg^{2+}} \nu^{0.5} \quad (4)$$

where $I_{peak}$ is the peak current, $A$ is the contact surface area of the electrochemically active material, $n$ is the number of electron equivalents per reaction species, $c$ is the $Mg^{2+}$ concentration in the aqueous electrolyte, and $\nu$ is the scan rate. As shown in Fig. 6d, the cathode $I_{peak}$ (C1) and anode $I_{peak}$ (A1) show a linear relationship with the square root of $\nu$ ($\nu^{0.5}$), implying diffusion-controlled cell kinetics. The values of the A and C peaks of the $Mo_6S_8$//Mg–Sn@Mg cell calculated from the linear fitting slope are $4.53 \times 10^{-9}$ and $4.20 \times 10^{-9}$ $cm^2$ $s^{-1}$ (Table S5†), respectively. The values of the A and C peaks of the $Mo_6S_8$//Mg–Bi@Mg cell are $5.51 \times 10^{-9}$ and $4.72 \times 10^{-9}$ $cm^2$ $s^{-1}$, respectively. Both of these values exceed the values of $Mo_6S_8$//bare Mg (the values of peaks A and C are $3.20 \times 10^{-9}$ and $2.75 \times 10^{-9}$ $cm^2$ $s^{-1}$, respectively). The high $D_{Mg^{2+}}$ of the full cells of Mg–Sn@Mg and Mg–Bi@Mg accelerates the transmission speed of $Mg^{2+}$. Similarly, the $Mg^{2+}$ ion transport rate of $Mo_6S_8$//Mg–Cu@Mg is much higher than that of bare Mg, as shown in Fig. S17.† Subsequent EIS tests on the resistance of the full cells (Fig. S18 and Table S6†) show that the $R_{ct}$ values of the Mg–Sn@Mg and Mg–Bi@Mg full cells are much lower than that of the bare Mg cells.

As shown in Fig. 6e, $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg full cells show excellent rate performance at gradually increasing current density. The $Mo_6S_8$//Mg–Sn@Mg full cell

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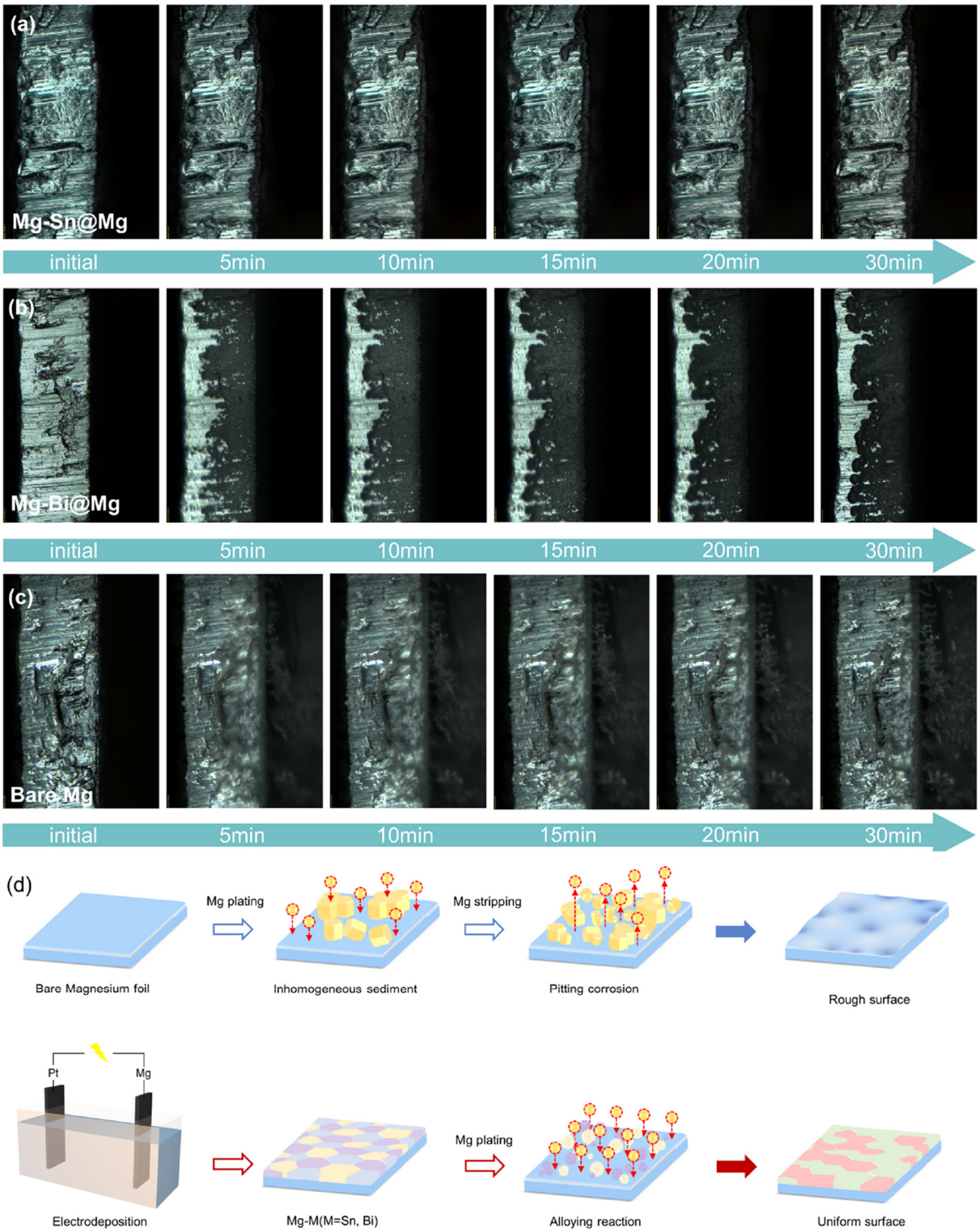


**Fig. 5** (a–c) *In situ* optical visualization observations of the Mg deposition. (d) Schematics of the electrochemical behavior of bare Mg and Mg–M@Mg (M = Tin and Bismuth) anodes.

shows specific capacities of 107.4, 94.1, 81.2, 72.0, 63.9 and 52.8 mA h $g^{-1}$ at different rates of 0.1, 0.2, 0.5, 1, 2 and 5C, respectively (Fig. S19†). The specific capacities recorded for $Mo_6S_8$//Mg–Bi@Mg (Fig. S20†) at the same current densities are 98.8, 90.4, 78.5, 68.4, 60.1 and 49.4 mA h $g^{-1}$, respectively. Both $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg are much better than $Mo_6S_8$//bare Mg (specific capacities of 96.1, 83.6, 68.4, 56.5, 47.8 and 37.6 mA h $g^{-1}$ at different rates of 0.1, 0.2,

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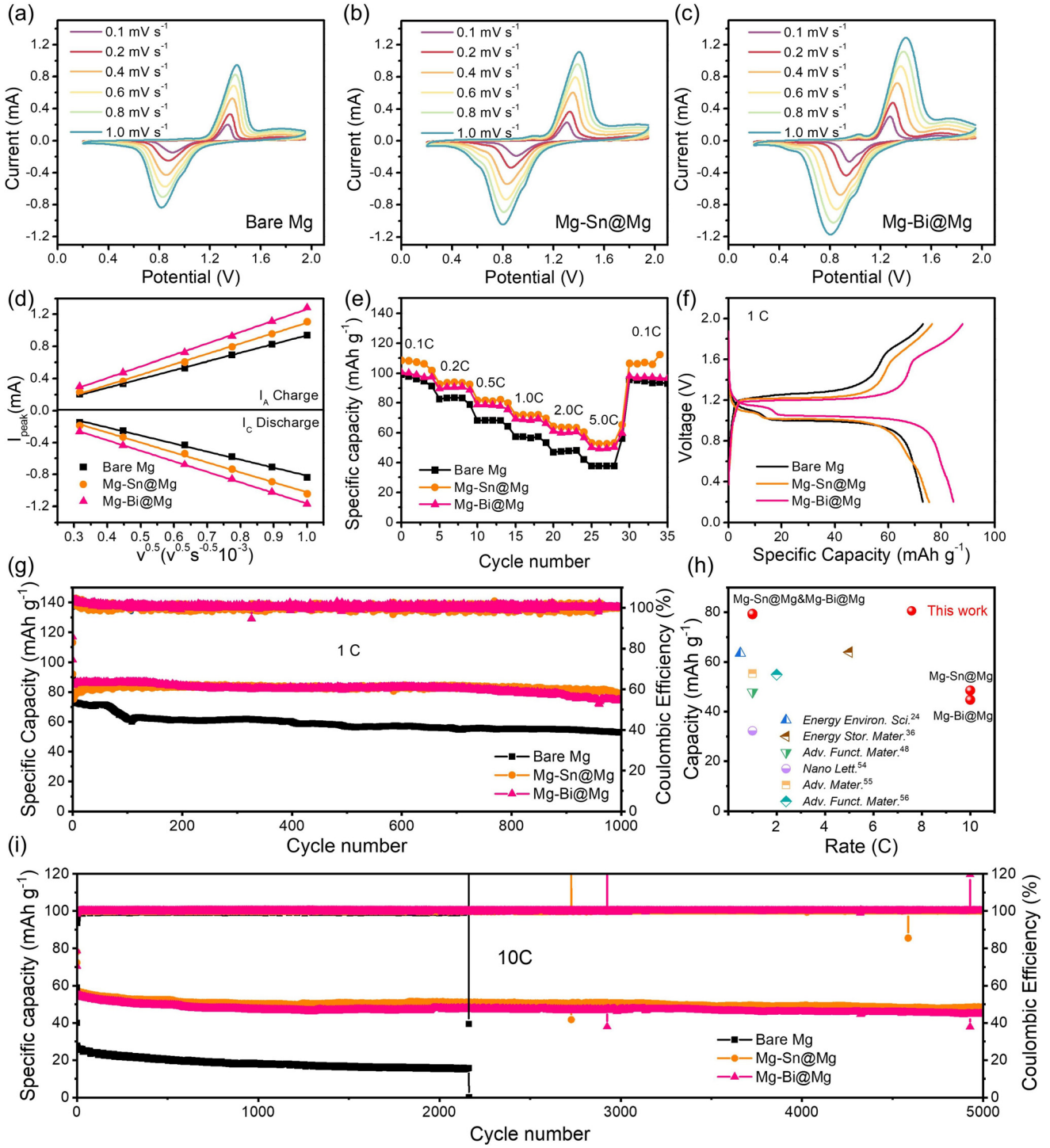


**Fig. 6** CV curves of (a) $Mo_6S_8$//bare Mg, (b) $Mo_6S_8$//Mg–Sn@Mg and (c) $Mo_6S_8$//Mg–Bi@Mg full cells at different scan rates from 0.1 to 1.0 mV $s^{-1}$ in the voltage range of 0.2–1.95 V and (d) the square root of scan rate ($v^{0.5}$) of bare Mg, Mg–Sn@Mg and Mg–Bi@Mg full cells. (e) Rate performance of the $Mo_6S_8$//bare Mg, $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg full cells. (f) The first charge–discharge profiles for $Mo_6S_8$//bare Mg, $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg full cells at 1 C. (g) Cycling performance of the $Mo_6S_8$//bare Mg, $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg full cells at 1 C. (h) The capacity and cycling stability performance of the $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg cells compared with those of previously reported $Mo_6S_8$//Mg full cells.[26,38,55,61–63] (i) Cycling performance of $Mo_6S_8$//bare Mg, $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg full cells at 10 C.

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0.5, 1, 2 and 5C, respectively) (Fig. S21†). The excellent multiplicative performance of $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg full cells is attributed to the Mg deposition kinetics associated with the enhanced diffusion rate of $Mg^{2+}$ ions.

In addition to the excellent rate performance, the $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg cells also possess extraordinary cycling stability. The first charge–discharge profiles at 1 C are shown in Fig. 6f, and the capacities of both Mg–Sn@Mg and Mg–Bi@Mg are higher than that of bare Mg. At a current density of 1 C (Fig. 6g), $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg cells can deliver high discharge capacities of 79.2 and 74.6 mA h $g^{-1}$ after 1000 cycles, respectively. $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg full cells have significant capacity retention rates of 85.9% and 87.0%, corresponding to the average Coulomb efficiency of 99.79% and 99.86%, respectively. Moreover, the $Mo_6S_8$//Mg–Cu@Mg cell also exhibits excellent cycling stability, with a high specific capacity of 81.8 mA h $g^{-1}$ after 700 cycles (Fig. S22†). As shown in Fig. S23,† the $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg full cells exhibit stable performance and good capacity retention at 0.5 C. Such cycling performance of full cells using Mg–Sn@Mg and Mg–Bi@Mg exceeds that of other full cells using $Mo_6S_8$ as cathodes (Fig. 6h), featuring the merits of Mg–Sn@Mg and Mg–Bi@Mg. At a high current rate of 10C (Fig. 6i), the $Mo_6S_8$// bare Mg cell shows rapid capacity decay over 2163 cycles. Notably, the initial discharge capacity of the $Mo_6S_8$// bare Mg cell is 40.0 mA h $g^{-1}$. However, the $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg-Bi@Mg batteries maintained high specific capacities of 48.6 and 44.8 mA h $g^{-1}$ after 5000 cycles.

## Conclusion

In summary, Mg–Sn and Mg–Bi alloy layers were successfully constructed on magnesium metal by a facile alloy electrodeposition strategy, which significantly solved the problems of slow $Mg^{2+}$ ion transfer kinetics and unstable anode/electrolyte interfaces, and thus improved the cycling performance of magnesium metal anodes. The as-prepared Mg–Sn@Mg and Mg–Bi@Mg symmetric cells exhibit ultra-stable cycling performance (~1000 h) at low current densities (0.1 mA $cm^{-2}$ and 0.2 mA $cm^{-2}$) and a higher current density (1.0 mA $cm^{-2}$). More importantly, compared with the $Mo_6S_8$//Mg full cell, the $Mo_6S_8$//Mg–Sn@Mg and $Mo_6S_8$//Mg–Bi@Mg full cells maintained an ultra-stable electrochemical performance even after 5000 cycles. This work provides a simple and low-cost strategy to optimize the Mg metal anode interface, which would be a promising direction for the design of high-performance RMBs and it may shed light on the development of next-generation rechargeable multivalent metal batteries.

## Author contributions

H. T. conceived the idea for the project and conducted the project. X. C., Y. X., F. Z., and B. H. wrote the manuscript. H. X. helped X. C. prepare alloy anodes in this work. All the authors agreed upon the final version of the manuscript.

## Conflicts of interest

The authors declare no competing financial interest.

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