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**HIGHLIGHT**

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oxide polymorphs



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## Inclusion and intercalation compounds of arsenic(III) oxide polymorphs†

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Arsenic(III) oxycompounds exhibit rich structural chemistry due to the presence of stereoactive lone electron pairs on arsenic and secondary As...O bonds that stabilize variable conformations of chains and rings in the case of oxyanions and layers or molecules in the case of oxide. These two features are also responsible for the fact that arsenic(III) oxide forms inclusion and intercalation compounds. In this Highlight article, the discovery and characterization of molecular arsenic(III) oxide inclusion compounds with hydrogen and helium are presented. Also, structural studies of As<sub>2</sub>O<sub>3</sub> intercalation compounds with alkali metal halides and pseudohalides are thoroughly described. Finally, future research avenues in the field are presented, which have become more appealing due to the recent literature reports of extraordinary optical properties of arsenic(III) oxide intercalates.

### Introduction

The main building block of arsenic(III) oxide is a  $\psi$ -tetrahedron AsO<sub>3</sub> (see Fig. 1). The arsenic atomic core is surrounded by a stereoactive lone electron pair (LEP) and three oxide ligands which are shared with neighboring arsenic atomic cores so that

As<sub>2</sub>O<sub>3</sub> stoichiometry is achieved. As a consequence, arsenic(III) oxide forms a wide range of both crystalline and glassy polymorphs. They encompass layered forms comprising infinite (As<sub>2</sub>O<sub>3</sub>)<sub>∞</sub> sheets with varied degrees of corrugation and a molecular polymorph, arsenolite, with As<sub>4</sub>O<sub>6</sub> molecules that exhibit symmetry of the  $4\bar{3}m$  point group and are isostructural to the well-known P<sub>4</sub>O<sub>6</sub> molecules.<sup>1–7</sup>

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† Dedicated to the memory of Prof. Janusz Zachara (1955–2024) for his mentoring of the Author and contributions to the structural chemistry of arsenic.

However, the crystal structure of arsenolite is completely different from that of P<sub>4</sub>O<sub>6</sub> due to directional As...O interactions that are often dubbed secondary bonds.<sup>6,7,9</sup> They are located *trans* with respect to primary bonds and, similarly as hydrogen bonds in ice, cause arsenolite to be less densely packed than phosphorus(V) oxide.<sup>10</sup> Consequently, there are small voids in the crystal structure of arsenolite which, as will be presented in the article, leads to the propensity of arsenolite to form inclusion compounds with hydrogen and helium.

In the same vein, the crystal structures of layered As<sub>2</sub>O<sub>3</sub> polymorphs, naturally occurring claudetite I and synthetic claudetite II, are governed by the intra- and interlayer As...O interactions as well as mutual orientations of LEPs on neighboring arsenic atoms. The interlayer As...O interactions can be replaced with different interactions, which is the reason why As<sub>2</sub>O<sub>3</sub> forms numerous intercalation compounds in which charged layers of cations and anions are sandwiched between As<sub>2</sub>O<sub>3</sub> sheets.

The Highlight article is divided into two sections. The first one deals with arsenolite inclusion compounds with hydrogen and helium. Their discovery, as well as detailed investigations, are concisely presented. The second part concerns As<sub>2</sub>O<sub>3</sub> intercalates and starts with a historical introduction to their research. Then, the latest results in the field are presented. Instead of conclusions, future research avenues are proposed.



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**Fig. 1**  $\text{AsO}_3$   $\psi$ -tetrahedron (a); an  $\text{As}_4\text{O}_6$  molecule (b); two  $\text{As}_4\text{O}_6$  molecules from arsenolite crystal structure oriented in a face-to-face fashion with  $\text{As}\cdots\text{O}$  interactions depicted with brown dashed lines (c);<sup>1,8</sup> claudetite I and II crystal structures (d) and (e), respectively;<sup>2,3</sup> polymeric  $(\text{As}_2\text{O}_3)_\infty$  layers in claudetite I and II (f) and (g), respectively; schematic representation of  $(\text{As}_2\text{O}_3)_\infty$  layers in claudetite I and II (h) and (i), respectively, with  $\psi$ -tetrahedra depicted as triangles and the orientation of stereoactive LEPs on arsenic atomic cores directed toward opposite sides of the sheet indicated with white and grey. As and O atoms are depicted as green and red spheres, respectively. Reprinted from author's PhD thesis.

### Discovery and studies of arsenolite inclusion compounds with helium and hydrogen

During our high-pressure structural study of the crystal structure of arsenolite, helium was used as a pressure-transmitting medium (PTM).<sup>11</sup> It was found that arsenolite does not undergo polymorphic phase transitions up to at

least 30 GPa. Also, surprising deformations of  $\text{As}_4\text{O}_6$  molecules were observed. The distance of arsenic atomic cores from the  $\text{As}_4\text{O}_6$  cages' center of gravity increased, and, simultaneously, the distance of oxygen atomic cores from the cages' center decreased (see the bottom of Fig. 2).

The most exciting result of the research was an accidental observation that, starting from 3 GPa, there was a very weak





Fig. 2 The molecular structure of  $\text{As}_4\text{O}_6$  presented in the wireframe model and symmetry coordinates of molecular deformations: scaling  $\zeta$  (top) and distortion  $\eta$  (bottom). Arsenic and oxygen atoms denoted green and red, respectively.

reflection on the lower- $2\theta$  side in the vicinity of every reflection coming from an arsenolite single-crystal. After ruling out all possible instrumental reasons for the appearance of the weak reflections, a bold hypothesis was made that helium, which is usually treated as an inert PTM, can permeate the non-porous arsenolite structure and diffuse into small voids which are smaller than helium atoms. Consequently, the introduction of helium causes unit-cell expansion, and reflections stemming from the formed inclusion compound are located at lower values of diffraction angles than the corresponding arsenolite reflections. The intensity of the weak reflections coming from the inclusion compound, which we called clathrate at that time, was integrated, and its crystal structure was determined and refined (see Fig. 3). To our surprise, it was possible to find helium atoms in electron density difference maps and to determine the stoichiometry of the obtained compound to be  $\text{As}_4\text{O}_6 \cdot 2\text{He}$ .

This result was confirmed *via* quantum-mechanical computations within the density functional theory (DFT) framework with periodic boundary conditions. Equation of state (EoS), which is the unit cell volume dependence of the obtained inclusion compound on pressure, obtained from DFT calculations was in excellent agreement with the experimental EoS. In addition to that, the discovery was confirmed independently by a group of Spanish researchers who subjected powdered arsenolite to pressure in various PTM and observed the formation of the  $\text{As}_4\text{O}_6 \cdot 2\text{He}$  compound.<sup>13</sup>



Fig. 3 Crystal structure of arsenolite inclusion compounds with helium and hydrogen,  $\text{As}_4\text{O}_6 \cdot 2\text{He}$  and  $\text{As}_4\text{O}_6 \cdot 2\text{H}_2$ .  $\text{As}_4\text{O}_6$  molecules are presented in the wireframe model with As and O atoms denoted green and red, respectively. Helium atoms are presented as grey transparent spheres, whereas hydrogen molecules are shown in blue in the ball-and-stick model. Helium atoms and dihydrogen molecules are located at the 0, 0, 0 position. The most likely orientation of  $\text{H}_2$  molecules along  $\langle 111 \rangle$  crystallographic direction is shown. Adapted from P. A. Guńka, L. Zhu, T. A. Strobel and J. Zachara, *J. Chem. Phys.*, 2020, **153**, 054501,<sup>12</sup> with the permission of AIP Publishing.

The co-existence of the weak reflections coming from the arsenolite-helium inclusion compound with the reflections stemming from arsenolite was interpreted such that helium permeated only the surface layer of the studied single crystal, and there were two compounds present in the diamond anvil cell (DAC) during the experiment.<sup>11</sup> Subsequently, studies to elucidate the mechanism of helium permeation into non-porous arsenolite, to check whether it is possible to transform arsenolite into the inclusion compound with helium completely, and to reveal the thermodynamic driving force of the  $\text{As}_4\text{O}_6 \cdot 2\text{He}$  compound formation were undertaken. With this in view, a series of X-ray diffraction experiments were carried out on arsenolite single crystals under high pressure in helium and neon as PTM. They allowed for a conclusion that the mechanism of helium diffusion is two-fold. The first one consists most likely in the slow diffusion of helium atoms along temporary channels appearing in the arsenolite crystal due to crystal lattice vibrations (phonons).<sup>14</sup> The second mechanism dominates for low-quality crystals and consists in much faster helium diffusion along the borders of mosaic domains which real crystals are built of. The two postulated processes explain the fact that the complete transformation of an arsenolite crystal into the inclusion compound crystal lasts for 24 h under 5 GPa, causes crystal quality deterioration, and proceeds in such a way that two single-crystalline phases exist before the process is completed. However, decreasing pressure below 3 GPa causes practically instant reformation of arsenolite, and subsequent pressure increase causes the virtually instantaneous reformation of the inclusion compound.<sup>15</sup>

The thermodynamic driving force for the following reaction was revealed *via* DFT computations:





The idea behind the approach was to determine theoretically the free energy of all reagents at various pressures and to calculate the enthalpy of the above reaction by taking into account the  $pV$  term. Reaction enthalpy calculated in such a way equals Gibbs free energy at 0 K which is the thermodynamic potential at isobaric and isothermal conditions. In the above reaction equation, the phase of helium is denoted  $x$  because of a simplification that was made. The reaction takes place around 2–3 GPa, and helium is in a liquid phase at these pressures. It was assumed arbitrarily, though, that helium exists in a solid phase at all pressures, so periodic boundary conditions could be applied in all computations. Despite this approximation, the pressure above which the reaction of  $\text{As}_4\text{O}_6 \cdot 2\text{He}$  formation should be spontaneous is 1.8 GPa, according to the computations. It is in remarkable agreement with the experiment considering the above approximation and the fact that kinetic barriers were not taken into account. After modeling crystal structures of arsenolite and its inclusion compound with helium at numerous pressures, it was noticed that the introduction of helium atoms into arsenolite causes a decrease of  $\text{As}_4\text{O}_6$  molecular deformations at each pressure point which, as it turned out in further studies, leads to the reduction of conformational energy of the molecules. This observation inspired a decomposition of the enthalpy of reaction (1) into the following terms:  $p\Delta V$ ,  $E_{\text{def}}$ , and  $\Delta E_{\text{inter}}$ , which denote pressure-volume work, deformation energy of molecules due to the introduction of helium atoms into arsenolite structure, and the difference in the intermolecular interactions' energy in product and the sum of intermolecular interactions' energies in the substrates, respectively. This analysis concluded that the main driving force of reaction (1) is the  $p\Delta V$  term, and the second most significant factor is the deformation energy (see Fig. 4). It can be understood as the energetic gain arising from the change in  $\text{As}_4\text{O}_6$  molecular conformation due to repulsive interactions with helium atoms.<sup>15</sup>

A Spanish group of scientists led by Sans, who obtained the  $\text{As}_4\text{O}_6 \cdot 2\text{He}$  inclusion compound independently of us, concluded that there are specific  $\text{As} \cdots \text{He}$  interactions in its structure. They even claimed that they are localized bonds.<sup>13</sup> Such assertions seemed to be entirely untrustworthy from a chemical point of view to us, but, not to rely on chemical intuition alone, quantum-mechanical computations within the framework of symmetry-adapted perturbation theory (SAPT) were carried out. They showed beyond reasonable doubt that, at the  $\text{He} \cdots \text{As}_4\text{O}_6$  distances observed in the structure of the inclusion compound, the  $\text{He} \cdots \text{As}_4\text{O}_6$  interaction is repulsive and results mainly from the Pauli exclusion principle.<sup>15</sup>

The apparent contradiction that there is no net attraction between He atoms and  $\text{As}_4\text{O}_6$  molecules as suggested by SAPT analysis dominated by Pauli repulsion and the fact that  $\text{As}_4\text{O}_6 \cdot 2\text{He}$  is thermodynamically stable above 2 GPa shows an



Fig. 4 Enthalpy,  $p\Delta V$  term, and difference in energy of intermolecular interactions ( $\Delta E_{\text{inter}}$ ) for the  $\text{As}_4\text{O}_6 \cdot 2\text{He}$  formation reaction at 0 K together with  $\text{As}_4\text{O}_6$  deformation energy ( $E_{\text{def}}$ ) plotted as a function of pressure.  $p\Delta V_{\text{exp}}$  and  $p\Delta V_{\text{comp}}$  denote the  $p\Delta V$  term calculated using EoS fitted to the values of pressure and volume from the experiment and DFT computations, respectively. All energies are given per mole of  $\text{As}_4\text{O}_6$  molecules. Reprinted from P. A. Guńka, M. Hapka, M. Hanfland, M. Dranka, G. Chataśiński and J. Zachara, *ChemPhysChem*, 2018, **19**, 857–864,<sup>15</sup> with the permission of John Wiley and Sons. Copyright 2018, Wiley-VCH Verlag GmbH & Co. KGaA, Weinheim.

interesting principle that is currently the subject of scientific discussion. Namely, repulsive two-body contributions do not necessarily prevent the creation of new compounds or the stabilization of new phases as other contributions to Gibbs free energy may prevail over the local repulsions (see Fig. 4), *e.g.*  $pV$  term and long-range interactions in  $\epsilon$ -glycine.<sup>16</sup> Another contribution to Gibbs free energy that may favor the creation of new compounds is Madelung energy which was proven to drive the formation of inclusion compounds of helium with salts that have different stoichiometry from 1:1.<sup>17</sup> Introduction of helium atoms into such crystal structures alleviates the Coulomb repulsions of the majority ions that have the same charge. It was also shown rigorously that the same effect is the energetic driving force for the formation of  $\text{Na}_2\text{He}$  compound which is an electride with localized pairs of electrons.<sup>17,18</sup>

Obtaining the arsenolite inclusion compound with helium allowed one to suppose hydrogen should also form an analogous compound. Synchrotron X-ray diffraction experiments revealed that hydrogen forms, indeed, an analogous inclusion compound already at pressure  $\sim 1.5$  GPa. In this case, its crystal structure determination was trickier than in the case of the inclusion compound with helium. The difficulties were caused, on the one hand, by the lower quality of diffraction data and, on the other hand, by the fact that a dihydrogen molecule has more degrees of freedom than a helium atom. At this research stage, stoichiometry  $\text{As}_4\text{O}_6 \cdot 2\text{H}_2$  and a crystal structure in which  $\text{H}_2$  molecules occupy analogous sites as helium atoms and are aligned along crystallographic  $\langle 111 \rangle$  direction were proposed based on DFT computations (see Fig. 3). The assumptions on the same stoichiometry and quasiisostructurality of the arsenolite



## Highlight



Fig. 5 The ratio of hydrogen or helium molar volumes to the difference between molar volumes of arsenolite and its inclusion compound with hydrogen or helium plotted as a function of pressure.  $X = \text{H}_2, \text{He}$ . Reprinted with permission from P. A. Guńka, M. Hapka, M. Hanfland, G. Chataśiński and J. Zachara, *J. Phys. Chem. C*, 2019, **123**, 16868–16872.<sup>19</sup> Copyright 2019 American Chemical Society.

inclusion compounds with helium and hydrogen were made based on the observation that both compounds are formed at the pressure at which the ratio of hydrogen or helium molar volumes to the difference in molar volumes of arsenolite and its inclusion compound reaches the same value (see Fig. 5).<sup>19</sup>

Raman spectra of the compound  $\text{As}_4\text{O}_6 \cdot 2\text{H}_2$  were also recorded. A strong redshift of dihydrogen vibron (the band corresponding to rotational–vibrational transitions of  $\text{H}_2$  molecule) was observed, and it split into three components below 5 GPa, and two at higher pressures. This showed that the H–H bond is significantly weakened in the inclusion compound. DFT computations with periodic boundary conditions and subsequent analysis of crystal orbital Hamilton populations (COHP) and crystal orbital overlap populations (COOP) in the theoretically calculated electron density distribution revealed that the H–H bond becomes weaker and more polarized due to the interactions with arsenic lone electron pair (LEP).<sup>19</sup>

The observation of dihydrogen vibron splitting was surprising and puzzling because the diffraction data and DFT computations suggested the presence of an  $\text{H}_2$  molecule in one symmetry-independent position only. To understand this phenomenon, more Raman measurements on polycrystalline arsenolite samples in protium and deuterium under high pressure, both at room temperature and low temperatures down to 80 K, were carried out (see Fig. 6). These studies showed that the dihydrogen vibron is split into three components in the whole investigated pressure range (up to 20 GPa), and two out of the three vibrons stem from the *ortho* and *para* dihydrogen spin isomers which result from hydrogen nuclear spin. Unfortunately, the explanation of the cause of the third vibron was not possible based on these experiments.<sup>12</sup> The working hypotheses assuming a different inclusion compound stoichiometry or disorder in its structure consisting in some dihydrogen molecules occupying a different site in the crystal were ruled out with high likelihood *via* molecular dynamics modeling. The third

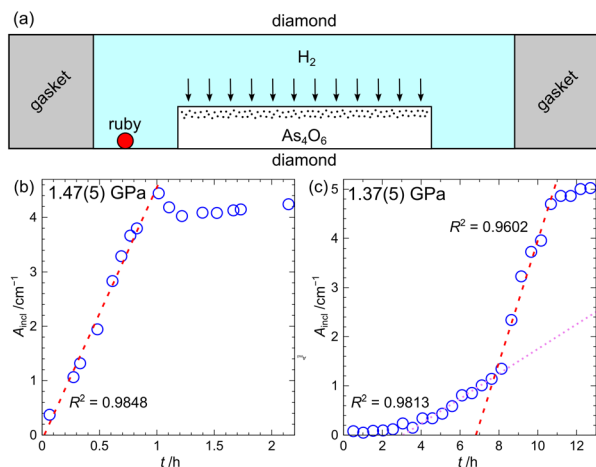


Fig. 6 Raman spectrum of  $\text{As}_4\text{O}_6 \cdot 2\text{H}_2$  inclusion compound in the wavenumber range corresponding to hydrogen vibrons recorded at 85(2) K and 1.29(5) GPa.  $Q_1(1)$  and  $Q_1(0)$  denote vibrons of free hydrogen, which was used as PTM, whereas  $\nu_1$ ,  $\nu_2$ , and  $\nu_3$  denote vibrons of  $\text{H}_2$  molecules present in the inclusion compound. Vertical lines represent pairs coming from *ortho* and *para* spin isomers of dihydrogen. Solid and dotted lines correspond to vibrons in which adjacent hydrogen molecules vibrate in and out of phase, respectively. Dashed lines indicate that the  $\nu_3$  vibron does not occur as a pair coming from two spin isomers, and its cause is different. Reprinted from P. A. Guńka, L. Zhu, T. A. Strobel and J. Zachara, *J. Chem. Phys.*, 2020, **153**, 054501,<sup>12</sup> with the permission of AIP Publishing.

vibron might result from more complex many-body interactions, and more research is necessary to elucidate its origin. The analysis of hydrogen rotors' shape, which are bands coming from hydrogen molecules being excited to higher rotational states, showed that the earlier hypothesis of total or, at least, partial suppression of  $\text{H}_2$  molecule rotations and its alignment along the  $\langle 111 \rangle$  direction is most likely valid, especially at low temperatures.<sup>12</sup>

Last but not least, the kinetics of hydrogen permeation into arsenolite single crystals have been determined using Fourier transform infrared (IR) spectroscopy at the SOLEIL synchrotron. Micrometer-thick  $\text{As}_4\text{O}_6$  single crystals have been obtained *via* spatial confinement by decomposing ammonium arsenite between glass slides.<sup>20,21</sup> The thin plates of arsenolite were placed in diamond anvil cells together with hydrogen, and pressure was increased to *ca.* 1.5 GPa. It was assumed that hydrogen would enter the crystals only through the top large face (see Fig. 7). The thickness of the arsenolite hydrogen inclusion compound layer was determined from the absorbance of its bands observed in the IR spectra recorded as a function of time. It was found that the growth of the  $\text{As}_4\text{O}_6 \cdot 2\text{H}_2$  inclusion compound was linear, indicating the process was not diffusion-limited, but the reactions taking place at phase boundaries were rate-limiting (see





**Fig. 7** Schematic drawing of the applied experimental setup; note that proportions between different parts in the drawing are not maintained (a). The peak area of inclusion compound  $\text{As}_4\text{O}_6 \cdot 2\text{H}_2$  plotted as a function of time for the peak around  $250\text{ cm}^{-1}$ . Measurements were carried out at 1.47 (5) and 1.37 (5) GPa (b and c, respectively). Figure adapted from P. A. Guřka, K. F. Dziubek and F. Capitani, *J. Phys. Chem. C*, 2023, **127**, 15871–15875.<sup>21</sup>

Fig. 7).<sup>21</sup> While only one region of linear growth was found at 1.47(5) GPa, two domains of linear growth with different slopes were observed at 1.37(5) GPa. The origin of this induction effect was not fully understood, but it was proposed that it may stem from the fact that, initially, islands of the inclusion compound are formed on the surface of arsenolite single crystal, and only after they have merged together does the layer of inclusion compound start to grow in thickness.<sup>21</sup>

### Studies of arsenic(III) oxide intercalates

As mentioned earlier, arsenic(III) oxide forms three ambient-pressure crystalline polymorphs. One of them, arsenolite, is a molecular solid, while the other two, claudetites I and II, are made of infinite  $\text{As}_2\text{O}_3$  layers. It was found a long time ago, much earlier than it was realized, that He atoms and  $\text{H}_2$  molecules can be inserted between  $\text{As}_4\text{O}_6$  molecules, that alkali metal halides can be introduced, *i.e.* intercalated, between the  $\text{As}_2\text{O}_3$  layers. According to the International Union of Pure and Applied Chemistry (IUPAC), intercalation compounds are substances formed due to reversible inclusion, without the formation of covalent bonds, of one kind of molecules into the solid matrix of a different compound with a layered structure.<sup>22</sup>

Arsenic(III) oxide intercalates are not typical examples of intercalation compounds as the introduction of ions between  $\text{As}_2\text{O}_3$  layers changes their conformation and involves breaking and re-formation of As–O covalent bonds. Several of them occur in nature as minerals.<sup>23–26</sup> Synthetic intercalates with potassium, ammonium, and sodium halides of a general formula  $n\text{MX} \cdot m\text{As}_2\text{O}_3$  were obtained for the first time in the 19th century, although their structure was unknown at that

time, and they were not considered intercalation compounds.<sup>27–30</sup> The crystal structure of the intercalates with the general formula  $\text{MX} \cdot 2\text{As}_2\text{O}_3$  was determined by Pertlik in the 1980s, though he used a different notation for their sum formula.<sup>31</sup> The compounds are built of  $\text{As}_2\text{O}_3$  sheets that separate alternating charged layers of cations and anions (see Fig. 8). The intercalates containing potassium and ammonium cations crystallize in the hexagonal  $P6/mmm$  space group and are isotypic. Hexagonal  $(\text{As}_2\text{O}_3)_n$  layers are flat, and all LEPS are pending on one side of the layers; the one which is adjacent to anions. The oxygen ligands are situated on the other side of the  $\text{As}_2\text{O}_3$  sheet coordinating cations (Fig. 8). The intercalates with such crystal structure will be denoted  $\text{P}_{\text{MX}}$  hereafter, where **P** denotes the crystal structure type determined by Pertlik and **MX** is the salt that intercalates arsenic(III) oxide. Contrary to other ammonium halides, ammonium chloride only forms a hydrated intercalate with a formula  $\text{NH}_4\text{Cl} \cdot \text{As}_2\text{O}_3 \cdot \frac{1}{2}\text{H}_2\text{O}$ , which also exhibits hexagonal symmetry and an analogous conformation of  $\text{As}_2\text{O}_3$  sheets.<sup>32</sup> A different structure has been observed for the intercalate with sodium bromide, which crystallizes in the orthorhombic crystal system, space group  $Pm\bar{c}n$ . The  $\text{As}_2\text{O}_3$  sheet is slightly more corrugated, and there are two kinds of  $\text{AsO}_2$  stripes within it: one with LEPS on all arsenic atoms pending to one side of the layers and another one with alternating LEPS (see Fig. 8).

Our studies of arsenic(III) oxide intercalates with potassium chloride have led to the crystallization of two new intercalates in addition to the known  $\text{KCl} \cdot 2\text{As}_2\text{O}_3$  ( $\text{P}_{\text{KCl}}$ ):  $\text{KCl} \cdot \text{As}_2\text{O}_3 \cdot 3\text{H}_2\text{O}$  ( $\text{X}_{\text{KCl}}$ ) and  $\text{KCl} \cdot \text{As}_2\text{O}_3 \cdot \frac{1}{2}\text{H}_2\text{O}$  ( $\text{Y}_{\text{KCl}}$ ). Intercalate  $\text{X}_{\text{KCl}}$  undergoes slow decomposition at room temperature to compound  $\text{Y}_{\text{KCl}}$ . The decomposition can be completed at 120 °C within half an hour. Heating intercalate  $\text{Y}_{\text{KCl}}$  to ~250 °C leads to intercalate  $\text{P}_{\text{KCl}}$  and free KCl. The crystal structures of both hydrated compounds have been determined, and it was found that they crystallize in the hexagonal crystal system and contain disordered water molecules (see Fig. 9). The crystal structure determination of intercalate  $\text{Y}_{\text{KCl}}$  was particularly interesting. Initially, the presence of water was overlooked. Thermogravimetric analysis coupled with mass spectrometry (TG-MS) and differential scanning calorimetry (DSC) allowed us to conclude that there is a mass loss of 3.19(5)% associated with an endothermic effect of  $174\text{ J g}^{-1}$  and the presence of water in gas in temperature range 150–180 °C. The presence of water was also confirmed by attenuated total reflection Fourier transform infrared (ATR-FTIR) spectroscopy and  $^1\text{H}$  NMR in the solid state. Interestingly, upon cooling from 200 °C, the intercalate re-absorbs moisture from the air, which results in the fact that powder diffraction patterns and ATR-FTIR spectra before and after heating are identical.<sup>34</sup> Analysis of unit cell parameters of compound  $\text{Y}_{\text{KCl}}$  and of compound  $\text{NH}_4\text{Cl} \cdot \text{As}_2\text{O}_3 \cdot \frac{1}{2}\text{H}_2\text{O}$ , whose structure was determined in the 1950s, together with the fact that other potassium–halide and ammonium–halide intercalates are isostructural allowed for a conclusion that the two compounds are also isostructural.<sup>34</sup> Consequently, a





**Fig. 8** Crystal structures of  $\text{As}_2\text{O}_3$  intercalates with potassium and sodium bromide.  $\text{KBr}\cdot 2\text{As}_2\text{O}_3$  ( $\text{P}_{\text{KBr}}$ ) crystal structure viewed down [102] and [001] directions (a) and (b), respectively;<sup>31</sup>  $\text{NaBr}\cdot 2\text{As}_2\text{O}_3$  structure viewed along [100] direction (c).<sup>33</sup> Schematic and ball-and-stick representations of polymeric  $(\text{As}_2\text{O}_3)_\infty$  layers in  $\text{NaBr}\cdot 2\text{As}_2\text{O}_3$  (d) and (e), respectively. The two kinds of  $\text{AsO}_2$  stripes mentioned in the text run horizontally.  $\psi$ -Tetrahedra are depicted as triangles, and the orientation of stereoactive LEPS on arsenic atomic cores directed toward opposite sides of the sheet indicated with white and grey. As, O, K, Na and Br atomic cores are depicted as green, red, violet, blue and brown spheres, respectively. Reprinted from author's PhD thesis.

corrected crystal structure model of compound  $\text{NH}_4\text{Cl}\cdot\text{As}_2\text{O}_3\cdot\frac{1}{2}\text{H}_2\text{O}$  was proposed and confirmed using X-ray diffraction and solid-state NMR studies.<sup>35</sup>

Subsequently, the first crystal structures of arsenic(III) oxide intercalates with rubidium cations were determined. A series of quasihydrothermal crystallizations of  $\text{As}_2\text{O}_3$  mixed with  $\text{RbCl}$  in molar ratios of 2:1, 1:1, and 1:2 in a solution having acidic, neutral, and alkaline pH were performed. Intercalate  $\text{RbCl}\cdot 2\text{As}_2\text{O}_3$  ( $\text{P}_{\text{RbCl}}$ ) isostructural with compound  $\text{KCl}\cdot 2\text{As}_2\text{O}_3$  ( $\text{P}_{\text{KCl}}$ ) crystallized from acidic solutions. In contrast, compound  $\text{RbCl}\cdot\text{As}_2\text{O}_3\cdot\frac{1}{2}\text{H}_2\text{O}$  ( $\text{Y}'_{\text{RbCl}}$ ) with a very similar structure to that of compound  $\text{KCl}\cdot\text{As}_2\text{O}_3\cdot\frac{1}{2}\text{H}_2\text{O}$  ( $\text{Y}_{\text{KCl}}$ )

precipitated from alkaline solutions. The only difference between the structures of these compounds occurs in the disordered layers. It is most likely related to the differences in ionic radii and, possibly, the acidity of  $\text{K}^+$  and  $\text{Rb}^+$  cations (see Fig. 10). Importantly, this difference also results in the existence of superstructure in compound  $\text{Y}'_{\text{RbCl}}$ , which was described using a supercell with a volume three times as large as that of the basic unit cell. Reflections indicating the presence of superstructure were only observed in the diffraction pattern stemming from a large single crystal of intercalate  $\text{Y}'_{\text{RbCl}}$ . Intercalates  $\text{P}_{\text{RbCl}}$  and  $\text{Y}'_{\text{RbCl}}$  were also characterized by  $^1\text{H}$  and  $^{87}\text{Rb}$  solid-state NMR spectroscopy.





**Fig. 9** Crystal structure of compounds  $P_{KCl}$  (a and d),  $Y_{KCl}$  (b and e), and  $X_{KCl}$  (c and f). Structures are presented in the ball-and-stick model along crystallographic [001] direction (a–c) and with thermal ellipsoids drawn at the 50% probability level along [120] direction (d–f). As, O, K, and Cl atoms are colored dark green, red, blue, and light green, respectively. The yellow color denotes the position of disordered potassium cations and water molecules. Hydrogen atoms were not refined in the structural models. Figure reprinted from M. A. Domański, K. Kraszewski, P. Paluch and P. A. Guńka, *Cryst. Growth Des.*, 2021, 21, 5215–5222.<sup>35</sup>



**Fig. 10** ATR-FTIR spectra of intercalates  $Y'_{RbCl}$  (solid black line) and  $Y_{KCl}$  (red dashed line). Inset above the legend shows the disordered layers located at  $z = \frac{1}{2}$ . The bottom right unit cells show the coordinates of the Wyckoff sites 3g and 6m in  $Y_{KCl}$  and  $Y'_{RbCl}$ , respectively, depicted with yellow spheres. In other unit cells possible random arrangement of cations and  $H_2O$  molecules is shown, taking into account their occupancies. As, O, K, Rb, and Cl atoms are depicted in dark green, red, dark blue, light blue, and light green, respectively. Figure reproduced from P. Michalak, P. Paluch and P. A. Guńka, *Cryst. Growth Des.*, 2022, 22, 711–717.<sup>36</sup>

Comparing the intercalates' spectra allowed for assigning signals in the  $^{87}Rb$  NMR spectrum of compound  $Y'_{RbCl}$  to rubidium cations occupying particular sites in its crystal structure.

Then, analysis of interlayer interaction energies in intercalate  $P_{RbCl}$  and isostructural intercalates with potassium halides  $P_{KX}$  was performed, and formation energies of these compounds from respective alkali-metal halides and arsenolite were

computed. This allowed us to conclude that the stability of intercalates with potassium halides increases with the increase of halide anion ionic radius. Similarly, the computations revealed intercalate  $P_{RbCl}$  is more stable than compound  $P_{KCl}$ .<sup>36</sup>

Last but not least, a comprehensive study of the anhydrous intercalates of potassium, rubidium, cesium, and ammonium halides ( $P_{MX}$ ) was carried out. For this purpose, the intercalates  $P_{RbBr}$ ,  $P_{RbI}$ ,  $P_{CsCl}$ ,  $P_{CsBr}$  and  $P_{CsI}$  were synthesized for the first



## Highlight

time. It was demonstrated that As...X interactions, where X is a halogen, are present in the intercalates, and they are analogous to the As...O secondary bonds that govern the structures of As<sub>2</sub>O<sub>3</sub> polymorphs and other As(III) oxycompounds. The As...X interactions are, therefore, also secondary bonds.<sup>9</sup> The analysis of As...X interactions was carried out using the analysis of arsenic LEP stereoactivity that was gauged using the bond-valence vector model proposed by Zachara.<sup>37</sup> The idea of this model is to assign a bond-valence vector to every bond formed by a coordination center that is parallel to the bond, directed from the coordination center to the ligand, and whose length is given by the following equation:

$$\|v_i\| = s_i \left(1 - \frac{s_i}{Q}\right)$$

where  $v_i$  and  $s_i$  denote bond-valence vector and valence of  $i$ th bond, respectively, whereas  $Q$  denotes coordination-center atomic core charge. For coordination spheres where no strain is present, the net bond-valence vector resulting from the summation of all bond-valence vectors should be a zero vector. If there is strain in a coordination sphere, the magnitude of the resultant vector is a measure of the strain magnitude. For example, if there is a stereoactive lone electron pair, *i.e.*, there is electronic strain, the net bond valence vector is large, and its length is a measure of lone-electron-pair stereoactivity. For a fully stereoactive LEP a valence  $s_{\text{LEP}}$  of 2 would be assigned and  $\|v_i\| = 2(1 - 2/5) = 1.2$  v.u.<sup>37</sup>

The strength of As...X interactions was assessed using the first-order valence entropy coordination number (<sup>1</sup>VECN).<sup>38</sup> Coordination number is one of the many terms used in chemistry that are very intuitive and broadly used but difficult to define precisely. For instance, in arsenic(III) oxycompounds, arsenic atomic cores form three strong, so-called primary bonds with oxygen atomic cores and, additionally, up to three weaker, so-called secondary bonds localized *trans* with respect to the primary bonds. In such cases, the arsenic coordination number is described as 3 +  $n$ , where  $n = 1, 2, 3$ . Such a description is not precise enough to compare subtle differences in the arsenic coordination sphere caused by secondary interactions or increasing pressure, among others. Bond-valence-based definitions of coordination numbers are ideally suited to such applications.<sup>38</sup> Here, only the definition of valence-entropy coordination number of order  $n$  (<sup>*n*</sup>VECN) will be presented so that readers can follow the analysis of the significance of As...X interactions. An interested reader is referred to ref. 38 to get acquainted with the definition of valence-diversity coordination number (VDCN). <sup>*n*</sup>VECN is defined by the following equation:

$$\left(\frac{\sum_{i=1}^N s_i^n}{k_{\text{VECN},n}}\right)^{\frac{1}{n}} = \lim_{\delta \rightarrow 0} \left(\frac{\sum_{i=1}^N s_i^{n+\delta}}{k_{\text{VECN},n}}\right)^{\frac{1}{n+\delta}}$$

where  $s_i$  denotes valency of the  $i$ th bond,  $k_{\text{VECN},n}$  stands for valence entropy coordination number of order  $n$  and  $N$  is the

number of bonds between coordination center and ligand which are taken into account. Relatively simple, but tedious, algebraic transformations allow one to get the following formula for the natural logarithm of <sup>*n*</sup>VECN:

$$\ln k_{\text{VECN},n} = - \sum_{i=1}^N \left(\frac{s_i^n}{S_n}\right) \ln \left(\frac{s_i^n}{S_n}\right)$$

where  $S_n = \sum_{i=1}^N s_i^n$ .

The application of these tools allowed for the analysis of the correlation between LEP stereoactivity and arsenic coordination number in the studied intercalates incorporating As...X interactions. The correlation overlain on the analogous correlation for As...O secondary bonds is presented in Fig. 11. The characteristic decrease in LEP stereoactivity correlated with an increase in <sup>1</sup>VECN

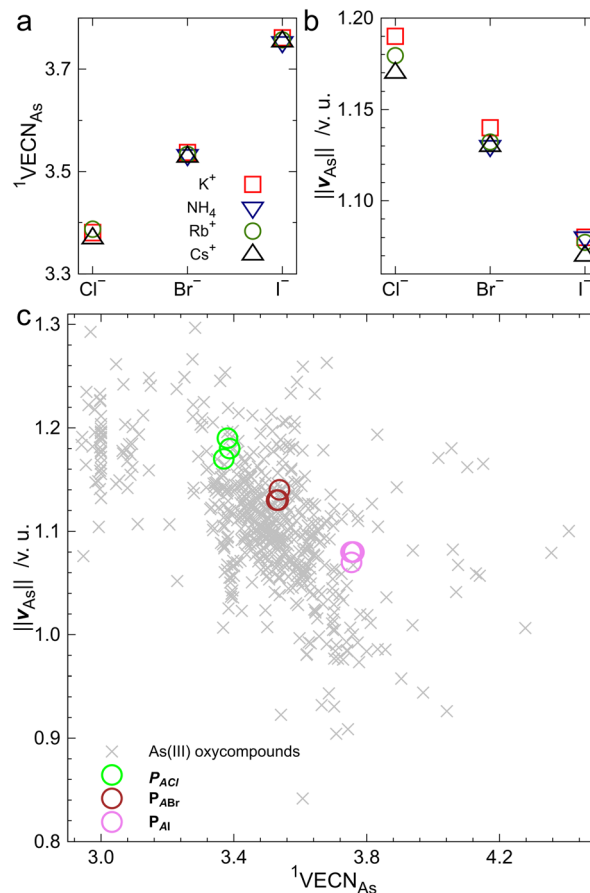


Fig. 11 Arsenic first-order valence entropy coordination number (<sup>1</sup>VECN<sub>As</sub>) and arsenic net bond valence vector length ( $\|v_{\text{As}}\|$ ), which equals the length of bond valence vector of arsenic lone electron pair, plotted as a function of the anion type present in the intercalates (a and b, respectively). <sup>1</sup>VECN<sub>As</sub> plotted as a function of  $\|v_{\text{As}}\|$  for arsenic(III) oxycompounds (denoted with grey x markers) and for anhydrous hexagonal P<sub>AX</sub> intercalates denoted with colored circles (c). Data for arsenic(III) oxycompounds taken from ref. 38. Figure reprinted from M. Dąbrowski, W. Wrzeźniewska, A. Prystupiak, J. Zachara and P. A. Guńka, *Cryst. Growth Des.*, 2023, 23, 6081–6085.<sup>39</sup>





Fig. 12 Comparison of potassium complex with 18-crown-6 and potassium neighborhood in  $P_{KX}$  intercalates where X stands for halide or pseudohalide anions. As, O, K, C, and H depicted as green, red, violet, grey, and white spheres, respectively.

together with the fact that  $As \cdots X$  interactions are located *trans* with respect to the primary As–O bonds testifies to the fact that  $As \cdots X$  interactions are indeed secondary bonds.

It is noteworthy that recently, an article appeared online that describes the first anhydrous polar  $As_2O_3$  intercalate  $CsBr \cdot As_2O_3$  ( $N_{CsBr}$ , where N indicates that the crystal structure of the intercalate is non-centrosymmetric) that exhibits a very strong second harmonic generation (SHG) response, which is 20.5 higher than that of  $KH_2PO_4$ . The intercalate  $P_{CsBr}$  studied in the same article exhibited very high birefringence (0.26), which was only slightly larger than that of  $N_{CsBr}$  (0.24).<sup>40</sup> Zeng and co-workers attribute the strong SHG effect to the fact that the polar  $AsO_3$   $\psi$ -tetrahedra are polymerized into  $(As_2O_3)_\infty$  layers and are all aligned in the same direction in  $N_{CsBr}$ . Actually, they designed the intercalates to attain such extraordinary optical properties.<sup>40</sup>

Last but not least, two intercalates with non-spherical pseudohalide azide anions have been obtained and structurally characterized:  $KN_3 \cdot 2As_2O_3$  ( $P_{KN_3}$ ) and  $NH_4N_3 \cdot 2As_2O_3$  ( $P_{NH_4N_3}$ ). Their structure is analogous to other  $P_{MX}$  intercalates where the linear azide anions replace halide anions and are oriented perpendicularly to  $As_2O_3$  sheets.<sup>41</sup> Detailed computational analysis of interlayer interaction energies between ions and arsenic(III) oxide layers, structural similarity of cations surroundings between  $As_6O_{12}$  rings to cations coordination by crown ether (see Fig. 12), structural similarity of potassium and ammonium cations, our earlier structural studies of ammonium oxoarsenates(III),<sup>42</sup> and the fact that intercalate with NaBr exhibits different conformation of  $As_2O_3$  layers allowed for a conclusion that alkali metal and ammonium cations template the conformation of  $As_2O_3$  layers and enforce their hexagonal symmetry described by the  $P6mm$  layer group.<sup>41,42</sup> Interestingly, comparative crystal structure analysis of the anhydrous intercalates  $P_{MX}$  revealed that the size of halide anions exerts a more significant influence on intercalate structural parameters, such as unit-cell parameters, As–O bond lengths,  $As \cdots X$  distances, and O–As–O and As–O–As valence angles, even though cations serve as templates for the  $As_2O_3$  layers and determine their conformation in the intercalates.<sup>39,41</sup>

## Conclusions

The great structural diversity of arsenic(III) oxycompounds is well represented in the inclusion and intercalation compounds of arsenic(III) oxide. It is caused by the fact that As(III) forms primary and secondary bonds. These weaker interactions not only stabilize various conformations in the  $As_2O_3$  polymorphs but are also responsible for the interactions with hydrogen, helium, and salts that co-form inclusions and intercalates with arsenic(III) oxide. There are still a lot of open questions concerning these compounds. For instance, the exact stoichiometry and structural model of the inclusion compound with hydrogen are questionable, and the reason for the third component of  $H_2$  Raman vibron is unclear. These issues could be resolved using neutron diffraction or inelastic neutron scattering experiments. Also, it remains an open question whether arsenic(III) oxide forms intercalates with other salts of sodium, lithium, and other cations, particularly ones with higher electric charge. I am convinced that pursuing these research questions will lead to new exciting results, and the recent paper by Zeng *et al.* describing the extraordinary optical properties of  $As_2O_3$  intercalates opens up a new perspective on the application potential of this class of compounds.

## Data availability

No primary research results, software or code have been included and no new data were generated or analysed as part of this highlight article.

## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 F. Pertlik, *Czech. J. Phys.*, 1978, **28**, 170–176.
- 2 F. Pertlik, *Monatsh. Chem.*, 1978, **109**, 277–282.
- 3 F. Pertlik, *Monatsh. Chem.*, 1975, **106**, 755–762.
- 4 K. S. Andrikopoulos, D. Christofilos, G. A. Kourouklis and S. N. Yannopoulos, *High Pressure Res.*, 2006, **26**, 401–406.



- 5 P. A. Guńka, M. Dranka, M. Hanfland, K. F. Dziubek, A. Katrusiak and J. Zachara, *Cryst. Growth Des.*, 2015, **15**, 3950–3954.
- 6 M. Jansen and M. Moebs, *Inorg. Chem.*, 1984, **23**, 4486–4488.
- 7 M. Jansen, M. Voss and H.-J. Deiseroth, *Angew. Chem.*, 1981, **93**, 1023–1024.
- 8 P. Ballirano and A. Maras, *Z. Kristallogr. – New Cryst. Struct.*, 2002, **217**, 177–178.
- 9 N. W. Alcock, in *Advances in Inorganic Chemistry and Radiochemistry*, ed. H. J. Emeléus and A. G. Sharpe, Academic Press, 1972, vol. 15, pp. 1–58.
- 10 F. Pertlik, *Monatsh. Chem.*, 1979, **110**, 387–392.
- 11 P. A. Guńka, K. F. Dziubek, A. Gładysiak, M. Dranka, J. Piechota, M. Hanfland, A. Katrusiak and J. Zachara, *Cryst. Growth Des.*, 2015, **15**, 3740–3745.
- 12 P. A. Guńka, L. Zhu, T. A. Strobel and J. Zachara, *J. Chem. Phys.*, 2020, **153**, 054501.
- 13 J. A. Sans, F. J. Manjón, C. Popescu, V. P. Cuenca-Gotor, O. Gomis, A. Muñoz, P. Rodríguez-Hernández, J. Contreras-García, J. Pellicer-Porres, A. L. J. Pereira, D. Santamaría-Pérez and A. Segura, *Phys. Rev. B*, 2016, **93**, 054102.
- 14 L. J. Barbour, *Chem. Commun.*, 2006, 1163–1168.
- 15 P. A. Guńka, M. Hapka, M. Hanfland, M. Dranka, G. Chałasiński and J. Zachara, *ChemPhysChem*, 2018, **19**, 857–864.
- 16 S. A. Moggach, W. G. Marshall, D. M. Rogers and S. Parsons, *CrystEngComm*, 2015, **17**, 5315–5328.
- 17 Z. Liu, J. Botana, A. Hermann, S. Valdez, E. Zurek, D. Yan, H. Lin and M. Miao, *Nat. Commun.*, 2018, **9**, 951.
- 18 X. Dong, A. R. Oganov, A. F. Goncharov, E. Stavrou, S. Lobanov, G. Saleh, G.-R. Qian, Q. Zhu, C. Gatti, V. L. Deringer, R. Dronskowski, X.-F. Zhou, V. B. Prakapenka, Z. Konôpková, I. A. Popov, A. I. Boldyrev and H.-T. Wang, *Nat. Chem.*, 2017, **9**, 440–445.
- 19 P. A. Guńka, M. Hapka, M. Hanfland, G. Chałasiński and J. Zachara, *J. Phys. Chem. C*, 2019, **123**, 16868–16872.
- 20 P. A. Guńka, M. Dranka, J. Piechota, G. Z. Żukowska, A. Zalewska and J. Zachara, *Cryst. Growth Des.*, 2012, **12**, 5663–5670.
- 21 P. A. Guńka, K. F. Dziubek and F. Capitani, *J. Phys. Chem. C*, 2023, **127**, 15871–15875.
- 22 *IUPAC Compendium of Chemical Terminology: Gold Book*, ed. M. Nič, J. Jiráč, B. Košata, A. Jenkins and A. McNaught, IUPAC, Research Triangle Park, NC, 3.0.1., 2009.
- 23 A. R. Kampf, B. P. Nash, M. Dini and A. A. M. Donoso, *Mineral. Mag.*, 2014, **78**, 747–755.
- 24 A. R. Kampf, B. P. Nash, M. Dini and A. A. M. Donoso, *Mineral. Mag.*, 2016, **80**, 1265–1272.
- 25 A. Garavelli, D. Mitolo, D. Pinto and F. Vurro, *Am. Mineral.*, 2013, **98**, 470–477.
- 26 U. Hålenius, F. Hatert, M. Pasero and S. J. Mills, *Mineral. Mag.*, 2016, **80**, 407–413.
- 27 H. Schiff and R. Sestini, *Justus Liebigs Ann. Chem.*, 1885, **228**, 72–91.
- 28 H. L. Wheeler, *Am. J. Sci.*, 1893, **46**(3), 88–98.
- 29 F. Rüdorff, *Ber. Dtsch. Chem. Ges.*, 1888, **21**, 3051–3053.
- 30 F. Rüdorff, *Ber. Dtsch. Chem. Ges.*, 1886, **19**, 2668–2679.
- 31 F. Pertlik, *Monatsh. Chem.*, 1988, **119**, 451–456.
- 32 M. Edstrand and G. Blomqvist, *Ark. Kemi*, 1955, **8**, 245–256.
- 33 F. Pertlik, *J. Solid State Chem.*, 1987, **70**, 225–228.
- 34 M. A. Domański, K. Kraszewski, P. Paluch and P. A. Guńka, *Cryst. Growth Des.*, 2021, **21**, 5215–5222.
- 35 W. Wrzeźniewska, P. Paluch and P. A. Guńka, *Acta Crystallogr., Sect. B*, 2023, **79**, 207–212.
- 36 P. Michalak, P. Paluch and P. A. Guńka, *Cryst. Growth Des.*, 2022, **22**, 711–717.
- 37 J. Zachara, *Inorg. Chem.*, 2007, **46**, 9760–9767.
- 38 P. A. Guńka and J. Zachara, *Acta Crystallogr., Sect. B*, 2019, **75**, 86–96.
- 39 M. Dąbrowski, W. Wrzeźniewska, A. Prystupiak, J. Zachara and P. A. Guńka, *Cryst. Growth Des.*, 2023, **23**, 6081–6085.
- 40 W. Zeng, Y. Tian, H. Zeng, Z. Lin and G. Zou, *Angew. Chem., Int. Ed.*, 2025, e202422818.
- 41 P. A. Guńka, K. Kraszewski, Y.-S. Chen and J. Zachara, *Dalton Trans.*, 2014, **43**, 12776–12783.
- 42 P. A. Guńka, M. Dranka and J. Zachara, *CrystEngComm*, 2011, **13**, 6163–6170.

