

## PAPER

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2, 1724Molybdenum sulfo-selenide nanocomposites with  
carbon nanotubes and reduced graphene oxide  
for photocatalytic hydrogen evolution reaction†Namsheer K,<sup>a</sup> K. Pramoda,<sup>\*a</sup> Kothanahally S. Sharath Kumar,<sup>b</sup>  
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We report a facile method for the preparation of nanocomposites of molybdenum sulfo-selenide ( $\text{MoS}_x\text{Se}_{1-x}$ ) with carbon nanotubes (CNTs) and reduced graphene oxide (RGO) via a solvothermal approach. The synthesized  $\text{MoSSe@CNT}$  and  $\text{MoSSe@RGO}$  nanocomposites are characterized by various microscopic and spectroscopic methods. We investigated the photocatalytic hydrogen evolution reaction (HER) performance of the  $\text{MoSSe@CNT}$  and  $\text{MoSSe@RGO}$  nanocomposites using eosin Y dye as a sensitizer, and triethanolamine as a sacrificial agent, under UV-vis light (Xenon lamp; 400 W) illumination. The highest photocatalytic HER activity of  $5016 \mu\text{mol h}^{-1} \text{g}^{-1}$  is achieved with the  $\text{MoSSe@RGO}$  nanocomposite, which is  $\sim 3$  times higher than that of the bare  $\text{MoS}_x\text{Se}_{1-x}$  nanostructure ( $1754 \mu\text{mol h}^{-1} \text{g}^{-1}$ ). Meanwhile, the  $\text{MoSSe@CNT}$  nanocomposite shows a somewhat lower activity of  $3622 \mu\text{mol h}^{-1} \text{g}^{-1}$ . These results indicate that the growth of  $\text{MoSSe}$  over conducting CNT/RGO improves the HER activity of the  $\text{MoSSe}$  nanostructure, but 2D RGO offers more facile electron-transfer routes compared to 1D CNTs. The HER activity of the nanocomposites is comparable with some of the recently reported 2D transition-metal-based nanocomposites reported in the literature. Photoluminescence studies indicate that the facile charge-transfer interaction between  $\text{MoSSe}$  and RGO/SWCNT is responsible for the remarkable HER activity of the nanocomposites. Additionally, more exposed edge sites of vertically grown discrete  $\text{MoSSe}$  nanostructures in the case of the nanocomposites, as suggested by microscopic studies, plausibly also contribute to the improved photocatalytic HER activity.

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## 1. Introduction

The generation of hydrogen by photocatalytic, electrocatalytic, and photo-electrocatalytic approaches has gained significant attention in recent years due to the rise in demand for renewable hydrogen.<sup>1–3</sup> Plants transform solar energy into chemical energy with a quantum yield of near unity wherein two photo-systems (photosystem I and photosystem II) concurrently absorb the light and in the cascade of electron-relay channels

help the spatial separation of charges.<sup>4,5</sup> Numerous efforts have been made to mimic natural photosynthesis in the laboratory artificially, to produce hydrogen and oxygen by splitting water.<sup>6–8</sup> A viable strategy for direct solar-to-hydrogen conversion has been reported to be photocatalytic water splitting over semiconductor materials without utilizing any fossil fuel or emitting carbon dioxide.<sup>9–11</sup> Thermodynamically, water splitting is an uphill reaction, and it involves a large Gibbs free energy change of  $237 \text{ kJ mol}^{-1}$ . Hence, several researchers have tried to generate hydrogen and oxygen from water using semiconductor photocatalysts to make water splitting viable.<sup>12–14</sup> Photo-induced water splitting on  $\text{TiO}_2$  electrodes was first reported by Fujishima and Honda in 1972.<sup>15</sup> Since then, several photosystems have been developed, which can imitate natural photosynthesis to generate hydrogen and oxygen from water by utilizing solar energy.<sup>16–18</sup> Photocatalysts utilized for water splitting range from narrow band gap to wide band gap materials.<sup>19–21</sup> Generally, narrow-band gap photosystems are preferred due to their comparatively higher visible-light absorption capacity. To achieve overall water splitting, the bottom of the CB of the semiconductor should be more negative than the  $\text{H}^+/\text{H}_2$  reduction potential

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(−0.41 V vs. NHE) and the top of the VB must be more positive than the water oxidation potential (+0.82 V vs. NHE).<sup>22,23</sup>

Since the discovery of graphene with novel properties,<sup>24,25</sup> two-dimensional (2D) transition metal dichalcogenides (TMDCs) have gained more importance due to their wide range of applications ranging from electronics<sup>26,27</sup> to catalysis.<sup>28,29</sup> The library of 2D TMDCs, including MoS<sub>2</sub>,<sup>30</sup> MoSe<sub>2</sub>,<sup>31</sup> WS<sub>2</sub>,<sup>32</sup> WSe<sub>2</sub>,<sup>33</sup> and MoTe<sub>2</sub>,<sup>34</sup> has been utilized in various photocatalytic/electrocatalytic applications due to their suitable physicochemical characteristics including large surface area, earth-abundant characteristics, and chemical inertness.<sup>35</sup> Normally, three processes are involved in photocatalytic water splitting: (1) light absorption by a catalyst, (2) charge-carrier generation/separation, and lastly (3) redox reactions at the catalyst's surface. The first criterion in the case of 2D materials is realized by tuning the bandgap either by phase engineering,<sup>36</sup> alloying,<sup>37</sup> defect engineering,<sup>38</sup> or heterostructuring,<sup>39</sup> while the thinning of the number of layers makes shorter travel distances for charge carriers to reach the surface active sites.<sup>22,23</sup> Finally, the third condition can be fulfilled by a high surface area of 2D materials, which offers a significant number of catalytically active sites.<sup>40</sup> On the other hand, bulk counterpart utilization is constrained in photocatalytic water splitting due to their poor light absorption, high charge-carrier recombination rate, and dearth of catalytically active sites.

Among the 2D TMDCs, MoS<sub>2</sub> is being extensively utilized as a catalyst for photocatalytic/electrocatalytic water reduction reactions under acidic as well as alkaline conditions because of the favourably located conduction band minimum (CBM), which is more negative than the water reduction potential (H<sup>+</sup>/H<sub>2</sub> 0 V vs. NHE) for feasible HER.<sup>41</sup> Structurally, individual MoS<sub>2</sub> layers consist of Mo metal atomic planes sandwiched between S atomic planes. Each monolayer is held together by weak van der Waals forces, forming a stacked structure.<sup>42,43</sup> The electrical and optical properties of MoS<sub>2</sub> can be tuned with a number of layers.<sup>43</sup> MoS<sub>2</sub> maintains various polymorphic forms, tetragonal (1T) and hexagonal (2H), depending on the procedure adopted for its synthesis. 1T-MoS<sub>2</sub> possesses a tetragonal crystal structure with octahedral (O<sub>h</sub>) coordination of the metal and chalcogen atoms while 2H-MoS<sub>2</sub> shows a hexagonal symmetry with trigonal prismatic (D<sub>3H</sub>) coordination. On Li-intercalation, 2H-MoS<sub>2</sub> is converted to a less stable 1T-phase.<sup>23,44</sup> The 1T-phase is reported to be metallic, while 2H-MoS<sub>2</sub> displays semiconductor characteristics. Due to their similar electronic configuration to Pt, theoretical models and experimental findings reveal that 2H-MoS<sub>2</sub> has tremendous potential for energy conversion applications.<sup>45</sup> However, the semiconducting characteristics and other stability issues associated with 2H-MoS<sub>2</sub> (hexagonal phase), which is the most stable form at room temperature, limit the application of MoS<sub>2</sub> as an HER catalyst.<sup>46</sup> Theoretical investigations revealed that the edge sites of 2H-MoS<sub>2</sub> are the only active sites for the water reduction reaction while the basal plane remains inert.<sup>47</sup> Combining 2D-MoS<sub>2</sub> with electrically conducting carbon nanotubes (CNTs) or graphene is reported to enhance the HER characteristics of bare TMDCs.<sup>48</sup> The growth of 2D TMDCs on

electrically conducting CNTs or graphene possibly reduces charge-transfer resistance and also increases the interaction between the electrolyte and the active sites of MoS<sub>2</sub>, since CNTs or graphene act as the template for the uniform growth of TMDCs.

Forming a solid-solution of MoS<sub>2</sub> with MoSe<sub>2</sub> is another interesting approach to enhance the MoS<sub>2</sub> HER activity as these solid-solutions show better charge-transfer characteristics, compared to individual components due to partial conversion of the semiconducting hexagonal 2H-phase to the metallic tetragonal 1T-phase.<sup>49</sup> Secondly, making solid solutions imports more active edge sites due to the formation of discrete MoS(Se) domains and also enhances dispersibility, thereby giving higher H<sub>2</sub> yield. Some of these MoS<sub>x</sub>Se<sub>1−x</sub> displayed stable hydrogen evolution under electrolytic conditions over a wide pH range.<sup>50</sup> However, a very limited number of studies focused on photocatalytic HER applications of these MoS<sub>x</sub>Se<sub>1−x</sub> solid solutions.<sup>51</sup> In addition, MoS<sub>x</sub>Se<sub>1−x</sub> solid-solution nanocomposites with CNTs or graphene are not reported for photocatalytic HER applications. Among various carbon supports, CNTs and graphene are considered good support materials because of their high electronic conductivity, large surface area, and ease of achieving surface modification, which helps to anchor active components.<sup>52,53</sup>

In the present study, bare MoS<sub>x</sub>Se<sub>1−x</sub> nanoflowers are synthesized by reacting ammonium molybdate tetrahydrate, thiourea, and selenium powder *via* a hydrothermal approach. Furthermore, the addition of carboxyl-functionalized multi-walled carbon nanotubes (MWCNTs) and graphene oxide (GO) independently during MoS<sub>x</sub>Se<sub>1−x</sub> nanoflower formation results in the growth of MoS<sub>x</sub>Se<sub>1−x</sub> over the CNT or graphene surface. The dye-sensitized water reduction activity of the synthesized nanocomposites is investigated. We find the highest H<sub>2</sub> evolution rate of 5016 μmol h<sup>−1</sup> g<sup>−1</sup> with the MoSSe@RGO heterostructure, whereas the MoSSe@CNT hybrid shows somewhat lower HER activity of 3622 μmol h<sup>−1</sup> g<sup>−1</sup>.

## 2. Results and discussion

Fig. 1 shows the schematic of the synthesis procedure employed to prepare the MoSSe@CNT and MoSSe@RGO heterostructures *via* the hydrothermal method. Initially, thiourea and selenium precursors (with hydrazine hydrate) decompose at high-temperature to give S<sup>2−</sup> and Se<sup>2−</sup> species, respectively, *via* hydrolyzation reaction as described in eqn (1)–(3). On the other hand, the ammonium molybdate precursor under reducing conditions and at high-temperature yields MoO<sub>3</sub> (eqn (4)).

Subsequently, the anion exchange reaction between MoO<sub>3</sub> and S<sup>2−</sup>/Se<sup>2−</sup> species yields a MoS<sub>x</sub>Se<sub>1−x</sub> intermediate. Finally, MoS<sub>x</sub>Se<sub>1−x</sub> is reduced to 1T and 2H-phases as the final product (eqn (5)).<sup>54</sup> The microstructure and surface morphology of the synthesized MoSSe@CNT and MoSSe@RGO nanocomposites are investigated using a field emission scanning electron microscope (FESEM). Bare MoSSe exhibits a flower-shaped morphology with vertically oriented petals on the spherical surface, as shown in Fig. (2a and d). After the addition of



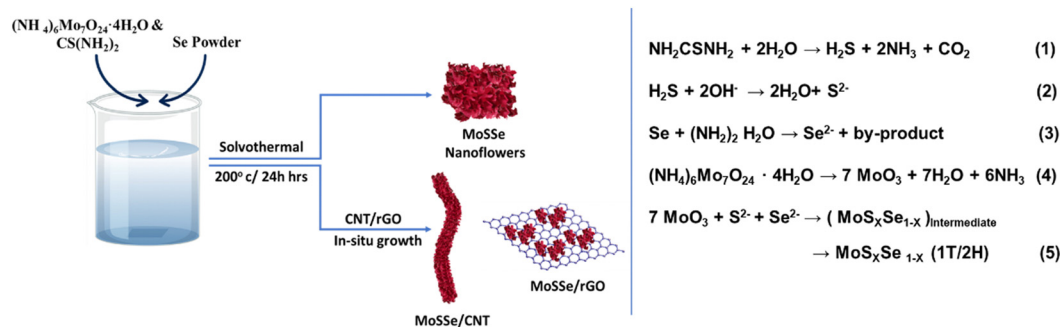


Fig. 1 A schematic illustration demonstrating the synthesis of bare MoSe, MoSe@CNT, and MoSe@RGO nanocomposites via a solvothermal approach and the right panel showing the various reaction intermediates generated during MoSe growth.

RGO/CNT, MoSe petals predominantly grown on 1D-CNT/2D-RGO are evident in the FESEM images shown in Fig. 2(b and e) and (c and f), which suggests that the functional groups of CNT/RGO act as templates for MoSe growth. Initially, positively charged molybdenum precursors get adsorbed onto the carboxyl-functionalized MWCNTs or RGO (negative zeta potential) surface by electrostatic interaction. Later, the reaction between molybdenum and sulfur/selenium precursors under hydrothermal conditions yields uniformly grown MoSe over CNTs or RGO. Fig. S1 (ESI<sup>†</sup>) shows the zeta potential plots for the functionalized SWCNTs, RGO, MoSe@CNT, and MoSe@RGO composites. The functionalized SWCNTs and RGO show negative zeta potential of  $-34$  and  $-32$  mV, respectively, due to the surface  $-\text{OH}$  and carboxyl groups. In the case of the MoSe@CNT ( $-8$  mV) and MoSe@RGO ( $-4$  mV) nanocomposites, the zeta potential shifted to a neutral value possibly due to electrostatic adsorption of MoSe precursors on RGO/CNT and later elimination of functionalities due to reducing conditions under hydrothermal conditions. Thus, zeta potential studies indicate that the surface functional groups of SWCNT/RGO act as templates for the growth of MoSe nanoflowers.

The defect-rich CNT/RGO templates perhaps minimize restacking/agglomeration of MoSe petals and result in the growth of isolated MoSe sheets over the CNT/RGO matrix. Formation of the hybrid structure provides enhanced active surface area and edge sites, leading to higher photocatalytic activity as compared to the bare MoSe nanostructures. Besides, the underlying electrically conducting CNT/RGO matrix improves the overall charge-transfer efficiency, and thereby the composites are expected to display higher photocatalytic activity. For comparison purposes, FESEM images of the carboxyl-functionalized MWCNTs and graphene oxide are shown in Fig. S2a and b (ESI<sup>†</sup>), respectively.

Fig. 3 shows transmission electron microscope (TEM) and high resolution TEM (HRTEM) images of the bare MoSe, MoSe@CNT and MoSe@RGO nanocomposites. The HRTEM images of MoSe@RGO and MoSe@CNT shown in Fig. 3b and d, respectively, revealed the direct growth of MoSe over the functionalized CNT/RGO surface. These results indicate that the surface functional groups of CNT/RGO act as anchor sites for nucleation and growth of MoSe nanostructures.

The powder X-ray diffraction pattern (PXRD) of the synthesized nanocomposites is recorded to investigate the phase

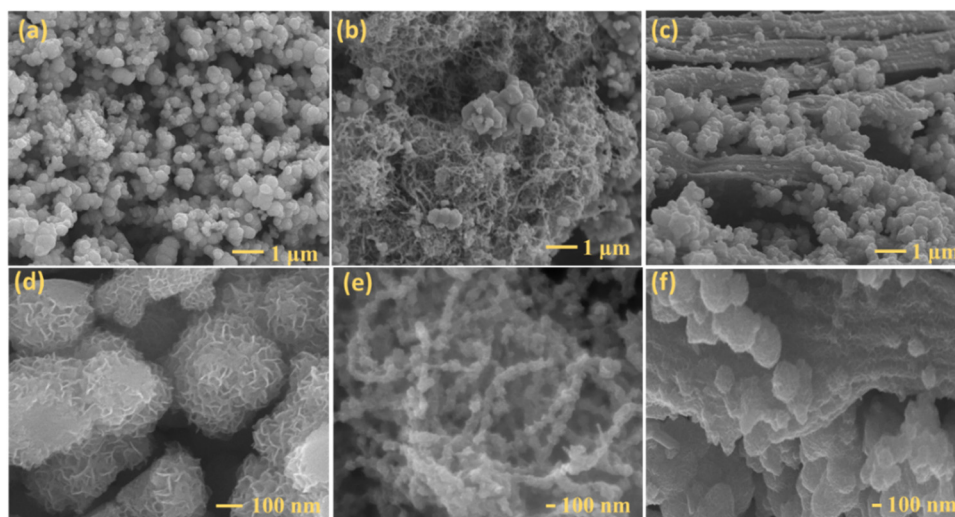


Fig. 2 Low and high-resolution FESEM images of (a) and (d) bare MoSe, (b) and (e) MoSe@CNT, and (c) and (f) MoSe@RGO heterostructures.



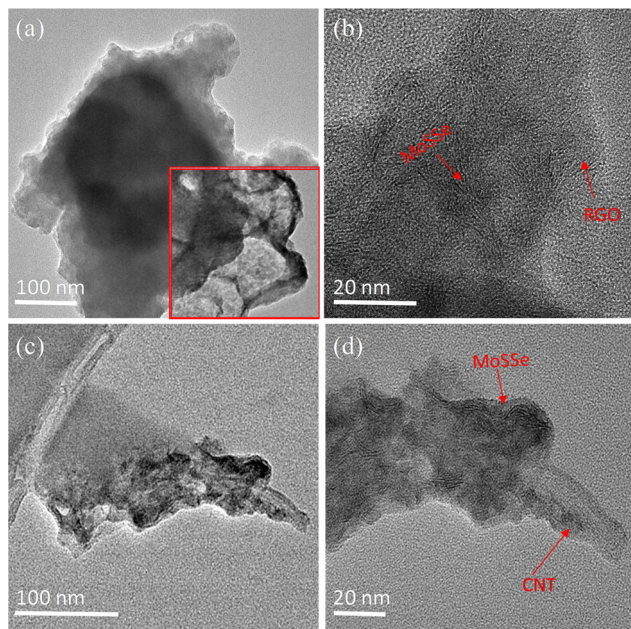


Fig. 3 Low and high-resolution TEM images of (a) and (b) MoSSe@RGO (inset shows TEM image of bare MoSSe), and (c) and (d) MoSSe@CNT heterostructures.

purity and other structural aspects. Fig. 4(a) shows the PXRD pattern of the MoSSe nanoflowers, MoSSe@CNT, and MoSSe@RGO respectively. Bare MoSSe nanoflowers show diffraction peaks at  $2\theta$  of  $9.2^\circ$ ,  $33.6^\circ$ ,  $43.6^\circ$ , and  $56.6^\circ$  attributed to (002), (100), (103), and (111) crystal planes, respectively, consistent with the JCPDS No: 37-1492 corresponding to 2H-MoS<sub>2</sub> nanosheets.<sup>55</sup> As per the literature, 1T/2H mixed phase MoS<sub>2</sub> shows the (002) plane of 1T and 2H-phases at  $9.0^\circ$  and  $14.0^\circ$ , respectively (JCPDS No. 37-1492). Bare MoSSe shows the main peak at  $2\theta$  of  $9.0^\circ$ , corresponding to the (002) plane of 1T-phase. While the peak at  $14.0^\circ$  attributed to the 2H-phase of the (002) plane is absent possibly due to the existence of a relatively higher proportion of the 1T-phase. However, the presence of

broad peaks at  $33.6^\circ$  and  $43.6^\circ$  indicates the presence of (110) and (103) planes of the 2H-phase,<sup>56</sup> indicating that the synthesized MoSSe is a mixture of 1T/2H-phase. In the case of the MoSSe@CNT and MoSSe@RGO nanocomposites, all the peaks corresponding to MoSSe are present along with a new peak at  $25.6^\circ$  corresponding to an inter-layer spacing of 0.35 nm between the stacked CNT/RGO evident, suggesting the growth of MoSSe over the CNT/RGO matrix.

Besides, we investigated the MoSSe@CNT and MoSSe@RGO nanocomposites by Raman spectroscopy to evaluate the MoSSe phase composition as well as the homogeneity of the nanocomposite (Fig. 4b) spectra.<sup>57,58</sup> Raman spectra recorded at various locations of the nanocomposite show peaks corresponding to both MoSSe and RGO/CNT, which confirms the homogeneity of the synthesized hybrids. The Raman spectra of the bare MoSSe display characteristic signals at 236, 283, and  $336\text{ cm}^{-1}$ , which are attributed to  $J_2$ ,  $E_{1g}$ , and  $J_3$  phonon modes of the 1T-phase. This result is in agreement with the report of Gupta *et al.*<sup>59</sup> Meanwhile, the peak at  $377\text{ cm}^{-1}$  corresponds to the  $E_{2g}^{1g}$  band of the 2H phase as reported by Jenjati *et al.*<sup>60</sup> In the case of the MoSSe@CNT and MoSSe@RGO heterostructure, the Raman peaks of the 1T/2H-phase are slightly red-shifted as compared to bare MoSSe, which suggests the charge-transfer interaction between the hetero-components. Meanwhile, the  $1343.5$  and  $1575\text{ cm}^{-1}$  peaks corresponding to the G and D bands of RGO,  $sp^3$  and  $sp^2$  hybridized carbon atoms, are also evident in the Raman spectra. X-ray photoelectron spectroscopy (XPS) is used to examine the phase purity and elemental composition of the MoSSe@RGO nanocomposite. The survey spectrum of the MoSSe@RGO nanocomposite shown in Fig. S3 (ESI<sup>†</sup>) displays the presence of only Mo, S, Se, and C elements, indicating that the nanocomposite synthesized is in a pure form and it is free from other impurities due to the starting precursors. The deconvoluted Mo 3d spectrum shows peaks at binding energies of 228.2 and 231.0 eV, which are attributed to the  $Mo^{4+} 3d_{5/2}$  and  $Mo^{4+} 3d_{3/2}$  components of MoSSe<sup>60,61</sup> (Fig. 5a). The other signals at 234.8 and 233.3 eV in the

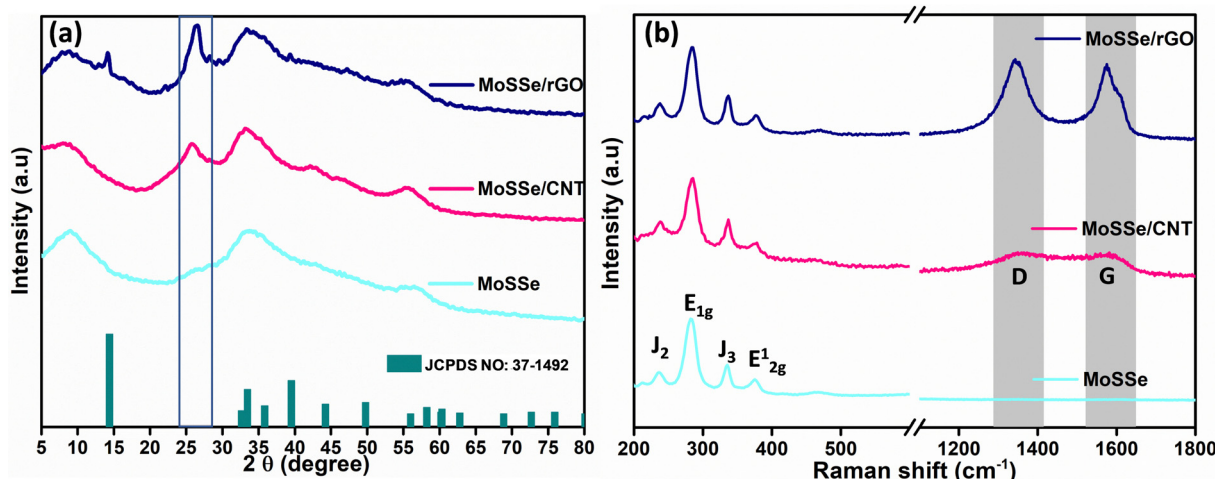


Fig. 4 (a) PXRD pattern and (b) Raman spectra of the bare MoSSe, MoSSe@CNT and MoSSe@RGO heterostructure.

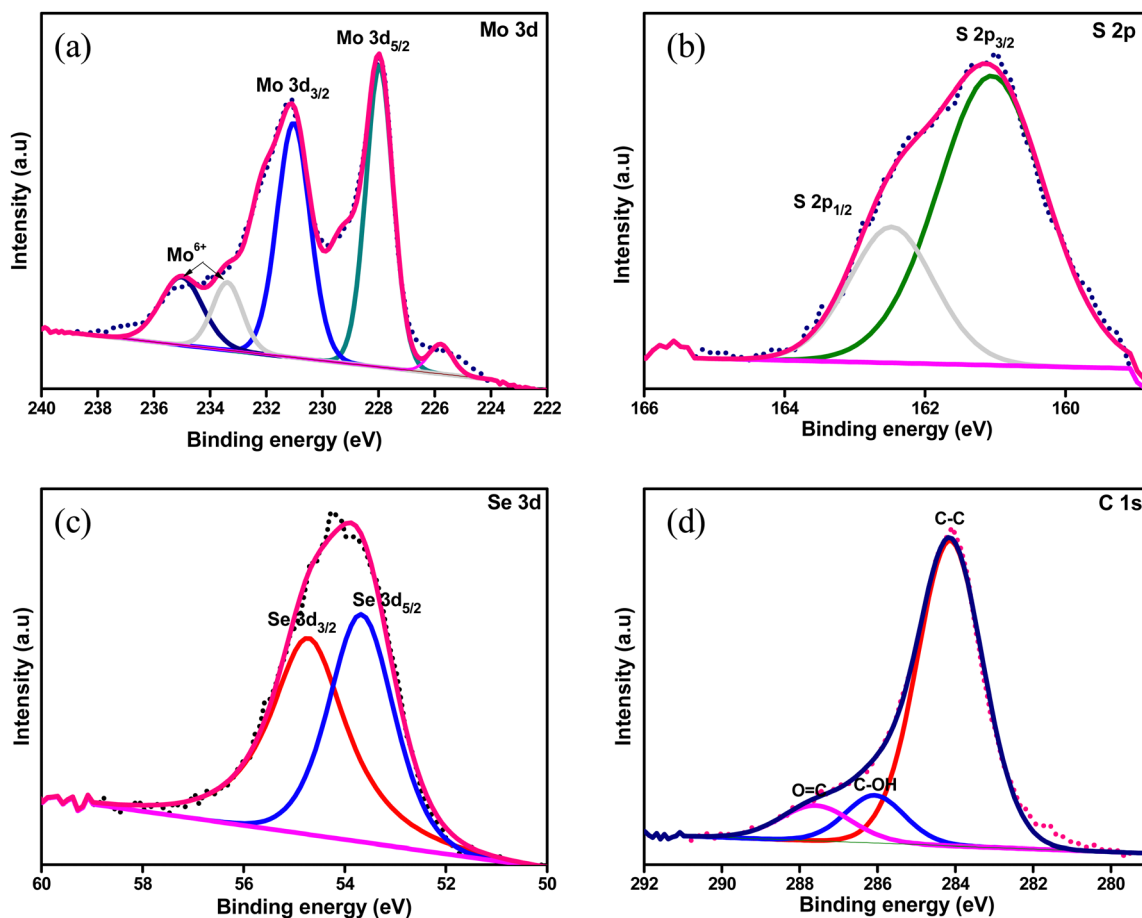


Fig. 5 High-resolution X-ray photoelectron spectra of (a) Mo 3d, (b) S 2p, (c) Se 3d and (d) C 1s elements of the MoSSe@RGO nanocomposite.

MoSSe@RGO nanocomposite are likely ascribed to the presence of a small portion of  $\text{Mo}^{6+}$ .<sup>62</sup> The deconvoluted S 2p spectrum shows peaks at 161.1 and 162.5 eV corresponding to the  $\text{S}^{2-}$  2p<sub>3/2</sub> and  $\text{S}^{2-}$  2p<sub>1/2</sub> state (Fig. 5b) respectively.<sup>63,64</sup> Similarly, the Se 3d spectrum of the MoSSe@RGO nanocomposite (Fig. 5c), displays two peaks at 53.7 and 54.7 eV corresponding to  $\text{Se}^{2-}$  3d<sub>5/2</sub> and  $\text{Se}^{2-}$  3d<sub>3/2</sub>, respectively.<sup>65</sup> Furthermore, the C 1s spectra are deconvoluted into three peaks. The peak at 284.6 eV corresponds to the graphitic carbon<sup>66</sup> and the peak at a binding energy of 286.0 eV is ascribed to hydroxyl groups and finally, the peak at 287.6 eV is attributed to the occurrence of a carbonyl group. The intensities corresponding to the oxygen functional groups, C–O and C=O, were negligible, suggesting the partial restoration of the sp<sup>2</sup> domains (Fig. 5d).<sup>67</sup> The XPS analysis confirms the presence of Mo, S, Se, and the reduced graphene matrix and the existence of a small proportion of Mo(vi) ions due to the presence of  $\text{MoO}_3$  by the incomplete reduction of  $\text{MoO}_4^{2-}$  during the reaction conditions.

The photochemical HER activity of the MoSSe@CNT and MoSSe@RGO nanocomposite is investigated using eosin Y dye as a photosensitizer and tri ethanol amine as a sacrificial agent under UV-light illumination. The yields of  $\text{H}_2$  evolved using the MoSSe@CNT and MoSSe@RGO nanocomposites along with the starting materials MoSSe, CNT, and RGO are shown in

Fig. 6(a and b). Pristine MoSSe shows HER activity of  $1754 \mu\text{mol h}^{-1} \text{g}^{-1}$  while the pristine SWCNTs and RGO display activity of 630 and  $651 \mu\text{mol h}^{-1} \text{g}^{-1}$ , respectively. In contrast, the MoSSe@CNT and MoSSe@RGO nanocomposites exhibit the activity of 3622 and  $5016 \mu\text{mol h}^{-1} \text{g}^{-1}$ , respectively. MoSSe@CNT shows a 2\* times enhancement in HER activity compared to the pristine MoSSe while MoSSe@RGO shows a slightly higher enhancement in activity of 3\* times. These results suggest that the growth of MoSSe nanostructures on the conducting CNT/RGO matrix enhances the HER activity. The functional groups of the underlying CNT or graphene networks act as nucleation sites for discrete MoSSe growth, which results in improved surface area and more exposed edge sites; thereby, the nanocomposites show higher HER activity.

The cycling stability tests revealed that MoSSe@RGO showed stable  $\text{H}_2$  evolution throughout 30 h and 6 cycles, indicating the robust nature of the nanocomposites (Fig. 6c). Furthermore, structural changes in the MoSSe@RGO nanocomposites after the cyclic stability test were investigated using Raman spectra. It can be seen that there were no noticeable changes in the Raman spectra even after the 30 h cycling stability test, as shown in Fig. S4 (ESI†). The apparent quantum yield (AQY) measured on the optimum MoSSe/RGO catalyst at 420, 500, 600 and 700 nm is included in Fig. 6d. The AQY for



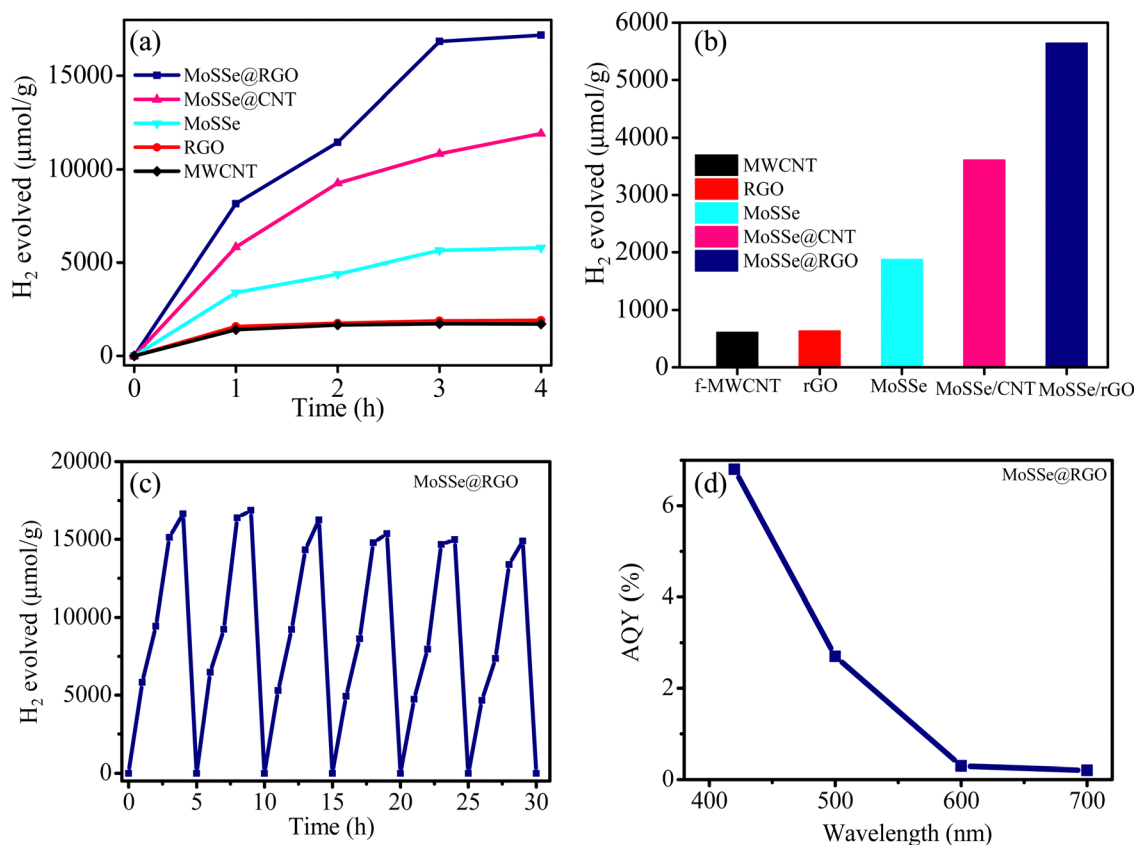


Fig. 6 (a) Photocatalytic H<sub>2</sub> evolution data of the bare MWCNTs, RGO, MoS<sub>2</sub>, MoS<sub>2</sub>@CNT, and MoS<sub>2</sub>@RGO nanocomposites; (b) bar diagram showing the comparison of the HER activities of the samples; (c) cycling stability curve for MoS<sub>2</sub>@RGO nanocomposites and (d) AQY measured on the optimum MoS<sub>2</sub>/RGO catalyst at 420, 500, 600 and 700 nm.

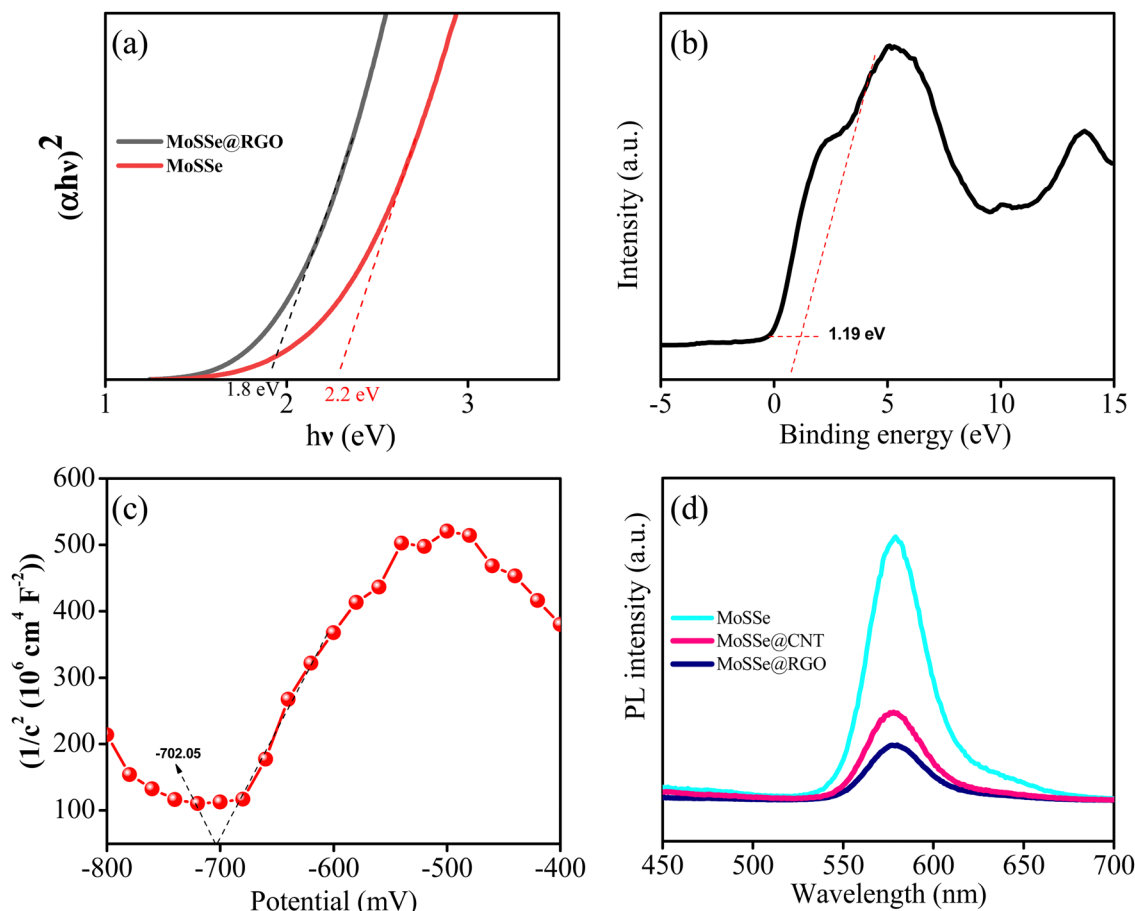
the MoS<sub>2</sub>/RGO catalyst is found to be 6.8% at 420 nm and 2.7% at 500 nm and it also exhibits some activity at 600 and 700 nm. In Table S1 (ESI<sup>†</sup>), we have compared the activity of the MoS<sub>2</sub>@CNT and MoS<sub>2</sub>@RGO nanocomposites with some of the other TMDC-based nanocomposites reported in the literature. The activity of the MoS<sub>2</sub>@RGO nanocomposites is found to be comparable with some of the best reported TMDC-based systems.

The band gap of the bare MoS<sub>2</sub> and MoS<sub>2</sub>@RGO is obtained from a Tauc plot derived from the UV-vis spectra (Fig. 7a). The Tauc plot is obtained by plotting  $h\nu$  vs.  $(\alpha h\nu)^2$ , where  $\alpha$  is the absorption coefficient,  $h$  is the Planck constant and  $\nu$  is the frequency of light. The obtained band gap energy value for the bare MoS<sub>2</sub> and MoS<sub>2</sub>@RGO nanocomposite is 2.2 and 1.8 eV, respectively. The band gap value of 1.8 eV obtained with MoS<sub>2</sub>@RGO indicates that the composite is suitable for photocatalysis application in visible-light radiation. To investigate the band positions, XPS valence band spectra (Fig. 7b) and Mott-Schottky plots (Fig. 7c) were explored. The valence band (VB) XPS spectra of MoS<sub>2</sub>@RGO revealed that the VB maximum of the MoS<sub>2</sub>@RGO nanocomposite is at 1.19 eV. Furthermore, the corresponding conduction band (CB) minimum was obtained using the following equation:

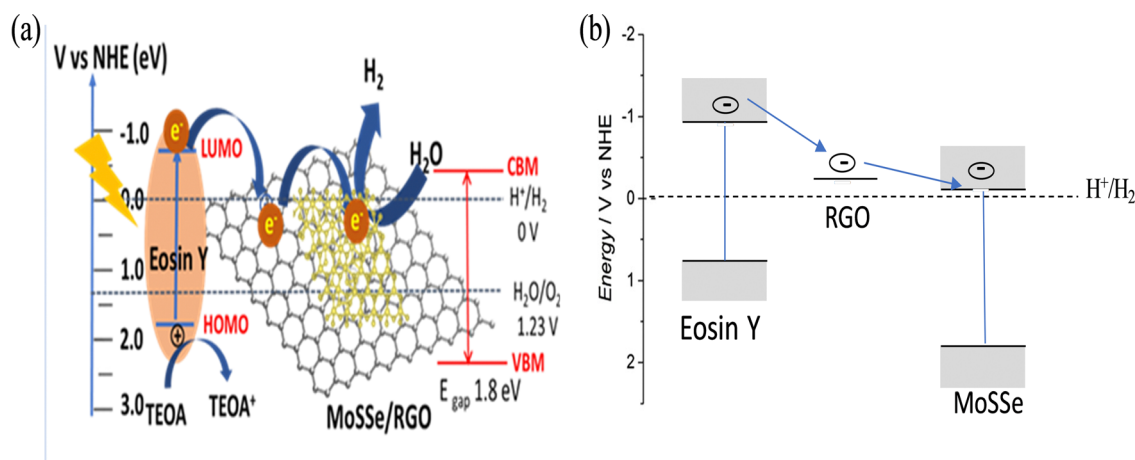
$$E_{CB} = E_{VB} + E_g$$

where  $E_{CB}$  and  $E_{VB}$  correspond to CB potential and VB potential, respectively, and  $E_g$  is the band gap. Therefore,  $E_{CB}$  of

MoS<sub>2</sub>@RGO is calculated to be at  $-0.61$  eV. Photoluminescence spectra (PL) of eosin Y dye are also examined in the presence of the MoS<sub>2</sub>@CNT and MoS<sub>2</sub>@RGO nanocomposites since the HER of all of the above mentioned samples is carried out in the presence of photosensitizer eosin Y dye. Bare eosin Y shows an emission peak at 570 nm in aqueous conditions when the excitation wavelength is set at 490 nm, consistent with the previous literature reports.<sup>68–70</sup> As shown in Fig. 7(d), the emission band of EY at 570 nm is quenched with the addition of the nanocomposites, suggesting the transfer of an electron from photoexcited eosin Y (EY\*) to the nanocomposites, which could possibly be utilized for the HER at the MoS<sub>2</sub> active sites. Based on the above studies, we have proposed a schematic for the mechanism of the HER on the MoS<sub>2</sub>@CNT and MoS<sub>2</sub>@RGO nanocomposites in the presence of Eosin Y dye (Fig. 8). On photoexcitation, EY undergoes a series of transformations to give EY<sup>3\*</sup> and then EY<sup>−1</sup>. The electron-rich EY<sup>−1</sup> later transfers electrons to the active sites of MoS<sub>2</sub>@CNT, namely the edge sites of MoS<sub>2</sub> *via* sp<sup>2</sup> bonded CNT/graphene networks, where water reduction takes place. Furthermore, a relatively lower magnitude of MoS<sub>2</sub> nanoflowers in the case of MoS<sub>2</sub>@CNT and MoS<sub>2</sub>@RGO compared to bare MoS<sub>2</sub> as evidenced by the FESEM images specifies that more edge sites are accessible in the nanocomposites, thereby showing higher activity. To further support the



**Fig. 7** (a) Tauc plot of MoSSe and the MoSSe@RGO nanocomposite; (b) VB XPS spectra and (c) Mott–Schottky plots of the MoSSe@RGO nanocomposite; (d) PL spectra of an aqueous solution of eosin Y ( $1.0 \times 10^{-7}$  M, excitation wavelength = 490 nm) in the presence of MoSSe@CNT and MoSSe@RGO.



**Fig. 8** (a) and (b) A schematic illustration representing the dye-sensitized photocatalytic HER mechanism of the MoSSe@RGO nanocomposites using TEOA sacrificial agent under visible-light irradiation. VBM (valence band minimum); CBM (conduction band maximum); HOMO (highest occupied molecular orbital); LUMO (lowest unoccupied molecular orbital).

elucidated mechanism, secondly, slightly higher enhancement in the case of MoSSe@RGO compared to MoSSe/SWCNT

suggests that graphene provides a more efficient pathway for electron transfer due to its 2D nature.



### 3. Conclusions

In conclusion, we have successfully synthesized MoSSe@CNT and MoSSe@RGO nanocomposites by a facile hydrothermal method. Zeta potential studies indicate that the functionalized CNT/RGO matrix acts as a template for the growth of MoSSe nanostructures. The synthesized composites are examined for the photocatalytic hydrogen evolution reaction using eosin Y as a photosensitizer and triethanolamine as a sacrificial agent, under UV-vis light. Both the nanocomposites show satisfactory activity towards H<sub>2</sub> evolution, the activities being 3622 and 5016  $\mu\text{mol h}^{-1} \text{g}^{-1}$  in the case of the MoSSe@CNT and MoSSe@RGO heterostructures, respectively. The HER activity of the nanocomposite is comparable with some of the 2D transition metal chalcogenide nanocomposites reported in the literature. The high HER activity of the MoSSe@CNT and MoSSe@RGO heterostructures as compared to bare MoSSe is attributed to the improved charge-transfer rate between the individual components and more exposed edge sites of MoSSe.

### Author contributions

Namsheer K: performing experiment, data interpretation, and writing the original draft. Dr K. Pramoda: performing experiments, conceptualization, editing the draft, funding acquisition, and supervision. K. S. Sharath Kumar: performing some portions of the experimental part. Prof. Chandra Sekhar Rout: conceptualization, project administration, funding acquisition, supervision.

### Conflicts of interest

The authors declare no conflict of interest.

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