

Cite this: *Nanoscale Adv.*, 2023, 5,  
1043

# Electrospun PVDF-based piezoelectric nanofibers: materials, structures, and applications

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Polyvinylidene fluoride (PVDF) has been considered as a promising piezoelectric material for advanced sensing and energy storage systems because of its high dielectric constant and good electroactive response. Electrospinning is a straightforward, low cost, and scalable technology that can be used to create PVDF-based nanofibers with outstanding piezoelectric characteristics. Herein, we summarize the state-of-the-art progress on the use of filler doping and structural design to enhance the output performance of electrospun PVDF-based piezoelectric fiber films. We divide the fillers into single filler and double fillers and make comments on the effects of various dopant materials on the performance and the underlying mechanism of the PVDF-based piezoelectric fiber film. The effects of highly oriented structures, core-shell structures, and multilayer composite structures on the output properties of PVDF-based piezoelectric nanofibers are discussed in detail. Furthermore, the perspectives and opportunities for PVDF piezoelectric nanofibers in the fields of health care, environmental monitoring, and energy collection are also discussed.

Received 4th November 2022  
Accepted 17th January 2023

DOI: 10.1039/d2na00773h

rsc.li/nanoscale-advances

## 1. Introduction

Piezoelectric materials have been extensively employed to build electronic skins, sensors, energy harvesters, *etc.* for applications in biomedical engineering, intelligent wearable electronics, and aerospace. Due to fluidity, low density, low cost, flexibility, environmental friendliness, and biocompatibility,<sup>1–3</sup> polyvinylidene fluoride (PVDF) as one of the most widely used piezoelectric polymer candidates attracts considerable research interest and has been applied in the fields of energy harvesting,<sup>4,5</sup> physiological health monitoring,<sup>6,7</sup> and environmental monitoring.<sup>8</sup> The PVDF material mainly refers to the homopolymer of vinylidene difluoride or the copolymer of vinylidene

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sensors for smart wearable products and health, and filtration materials for the environment.



difluoride and a small amount of other vinylidene monomers containing fluoride.

The primary factor influencing the piezoelectricity of PVDF is its crystalline phase, which has five crystalline varieties:  $\alpha$ ,  $\beta$ ,  $\gamma$ ,  $\delta$  and  $\epsilon$ .<sup>9,10</sup> Among them, the  $\beta$  phase has a planar zigzag all-trans conformation (TTT) with the  $-\text{CF}_2-$  group and the  $-\text{CH}_2-$  group on either side of the molecular chain, and the dipole moments are placed parallel and perpendicular to the  $c$ -axis.<sup>11–13</sup> Therefore, this conformation is the most polar conformation with the largest electric dipole moment and the maximum electroactivity.<sup>14</sup> Other than the  $\beta$  phase, a recent study reported that the highly mobile oriented ambient fraction (OAF) enables biaxially oriented PVDF films containing pure  $\beta$  crystals to obtain high voltage electrical properties.<sup>15</sup> However, most papers on electrospun PVDF fiber films still discuss the influence of the  $\beta$  phase on piezoelectric properties.

Different techniques, including stretching,<sup>16</sup> thermal annealing,<sup>17</sup> high-voltage electric field polarization,<sup>18</sup> as well as filler doping<sup>19</sup> and structural design,<sup>20</sup> have been utilized to induce

dipole alignment to enhance the  $\beta$ -phase proportion in PVDF. The stretching and thermal annealing processes are complicated and only slightly improve the piezoelectric output of pure PVDF when compared to other approaches. Polarization under a high electric field is indeed an effective method to improve the piezoelectric output of PVDF, but the limitation of the polarization direction leads to lack of flexibility in device design, and the preparation procedures are also complicated and time-consuming.<sup>21</sup> The methods of producing PVDF piezoelectric materials include tape casting,<sup>22</sup> coating methods,<sup>23</sup> thermal drawing,<sup>24–27</sup> electrospinning,<sup>5</sup> *etc.* However, the tape casting, coating, and thermal drawing also need post-treatment (such as high-voltage electric field polarization and thermal annealing) to make PVDF materials have better piezoelectric properties. In contrast, a quick and efficient electrospinning technique can create superfine fibers with diameters ranging from a few nanometers to a few microns and make it easy to incorporate different doping particles into the resultant PVDF nanofibers.<sup>28</sup> In addition, strong electric field force during the electrospinning



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process can facilitate the directional alignment of the  $-\text{CH}_2-/\text{CF}_2-$  dipoles and transform the  $\alpha$  phase into the  $\beta$  phase leading to the increase of the  $\beta$ -phase content in the PVDF nanofibers.<sup>29</sup> Furthermore, the formation of fiber films with different structures through various electrospinning processes can also boost the piezoelectric capabilities of PVDF fiber films.

Recent review articles published by others on the topic have focused on the impact of spinning parameters or one filler doping on the piezoelectric properties of electrospun PVDF films. This review will discuss the improvement of piezoelectric properties of PVDF-based fiber films through material doping and structural modification during the electrospinning process. The performance improvement of PVDF-based piezoelectric fiber films brought by doping various fillers and component combinations is first reviewed from the perspective of adding single and double fillers. Moreover, the impact of structural design on the performance of PVDF-based piezoelectric fiber films is discussed from the perspectives of high-orientation, core-shell, and multilayer structures. Finally, a summary of the current application state of electrospun PVDF-based piezoelectric fiber films is provided (Fig. 1).

## 2. Doping modification of PVDF nanofiber films

Some researchers tried to improve the piezoelectric properties of electrospun PVDF nanofiber films by altering solution

properties, environmental conditions, and spinning parameters. Zhao *et al.*<sup>30</sup> and Lei *et al.*<sup>31</sup> investigated the effects of solution concentration and solvent ratios on the  $\beta$ -phase content of PVDF fiber films, and the results showed that the fiber with a high  $\beta$ -phase content could be produced only when the solution concentration and solvent ratio are in the appropriate range. There were other articles that investigated the effect of PVDF molecular weight size on the piezoelectric properties of PVDF nanofiber films and found that the higher the molecular weight in a certain range, the higher the  $\beta$ -phase content.<sup>32</sup> In addition, several studies demonstrated that spinning parameters (applied voltage, feed rate, tip-to-collector distance, or diameter of a needle), ambient temperature, and humidity could also affect  $\beta$ -phase content, and the results showed that their most important role remained to regulate the spinnability of the polymer solution and the morphology of the nanofibers, with the spinning voltage and the tip-to-collector distance exhibiting partial contributions to the formation of the PVDF- $\beta$  phase.<sup>33–35</sup>

Recently, researchers further improved the piezoelectric properties of PVDF by incorporating specific concentrations of single or double fillers, such as piezoelectric ceramics and conductive particles, into the PVDF matrix to obtain composite nanofiber films.<sup>36–42</sup> It's worth noting that too large filler size will significantly increase the solution viscosity, which is not conducive to jet drafting in the electric field.

### 2.1 Single-filler doping

**2.1.1 Piezoelectric materials.** Piezoelectric materials include inorganic and organic piezoelectric materials. Inorganic piezoelectric materials include lead zirconate titanate (PZT),<sup>43</sup> barium titanate ( $\text{BaTiO}_3$ ),<sup>44,45</sup> sodium niobate ( $\text{NaNbO}_3$ ),<sup>46</sup> potassium sodium niobate (KNN),<sup>47</sup> zinc oxide ( $\text{ZnO}$ ),<sup>48,49</sup> *etc.* Organic piezoelectric materials include cellulose nanocrystals,<sup>50</sup> polyacrylonitrile (PAN),<sup>51,52</sup> nylon 11,<sup>53</sup> silk fibroin (SF),<sup>54</sup> *etc.* The addition of piezoelectric materials as a nucleating agent can not only increase the  $\beta$ -phase content of PVDF but also improve the crystallinity leading to the increase of the piezoelectric output. Moreover, the addition of inorganic nano materials in the matrix can enhance the strength of the polymer.<sup>55</sup>

PZT was the first widely used material due to its high piezoelectric coefficient ( $d_{33} = 140 \text{ pm V}^{-1}$ ).<sup>56</sup> It has been reported that electrospun PVDF/PZT nanofibers prepared by doping different concentrations of PZT could possess a higher  $\beta$ -phase content compared to pure PVDF. When PZT was supplied at a concentration of 10 vol%, the maximum effective voltage could reach 79.71 mV at a resistive load of 1 M $\Omega$  and 47.2 mV at a non-resistive load.<sup>57</sup> In addition, some researchers prepared PZT ceramic particles by combining sol-gel and electrospinning methods, and then electrospun PZT/PVDF nanocomposites containing different volume fractions of PZT were prepared and made into flexible pressure sensors. The findings demonstrated that the crystallinity and  $\beta$ -phase content of PVDF/PZT nanocomposites increased with the increase of PZT volume fraction, and the dielectric constant and

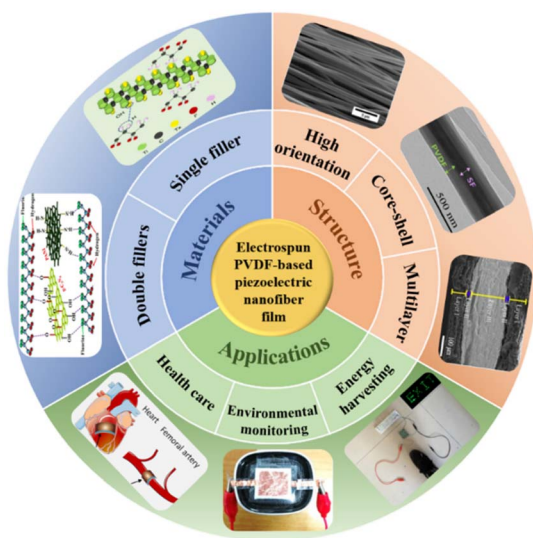


Fig. 1 Materials, structures, and applications of electrospun PVDF-based piezoelectric nanofiber films. (Go clockwise from double fillers: reproduced from ref. 131 with permission from the American Chemical Society, copyright 2019. Reproduced from ref. 132 with permission from Elsevier Ltd, copyright 2020. Reproduced from ref. 105 with permission from the MDPI, copyright 2018. Reproduced from ref. 109 with permission from the Chongqing University, copyright 2021. Reproduced from ref. 112 with permission from the Elsevier B.V., copyright 2020. Reproduced from ref. 67 with permission from the Elsevier B.V., copyright 2020. Reproduced from ref. 60 with permission from the American Chemical Society, copyright 2020. Reproduced from ref. 110 with permission from the American Chemical Society, copyright 2019).



piezoelectric coefficient also increased gradually. In comparison to pure PVDF nanofibers with an output voltage of 22 mV at a force of 2.4 N, PVDF/PZT nanocomposites had an output voltage of 184 mV at a force of 2.125 N when the volume fraction of PZT was 0.37 vol%.<sup>38</sup> Although PZT is the most widely used and effective piezoelectric material, it is unfavorable to the environment since it includes a lot of lead, which actually causes pollution.<sup>58</sup> Therefore, it has led to a growing interest in the use of lead-free piezoelectric ceramic materials as piezoelectric fillers to circumvent these restrictions.

As a highly effective lead-free piezoelectric ceramic material, BaTiO<sub>3</sub> ( $d_{33} = 130 \text{ pm V}^{-1}$ ) is widely employed.<sup>59</sup> Various PVDF/BaTiO<sub>3</sub> nanocomposites were fabricated with the electrospinning technique. The results showed that doping BaTiO<sub>3</sub> could increase the  $\beta$ -phase content of composites. The highest output voltage of piezoelectric nanogenerators (PENGs) made of PVDF nanofibers was 360 mV, whereas PENGs made of PVDF/BaTiO<sub>3</sub> nanocomposites with a 6% w/w concentration of BaTiO<sub>3</sub> had a peak output voltage of 740 mV.<sup>60</sup> In another study, the piezoelectric capabilities of electrospun PVDF/BaTiO<sub>3</sub> nanocomposites were enhanced by the addition of 5, 10, and 20 wt% of BaTiO<sub>3</sub> NPs.  $\beta$ -Phase content could reach 91.0% for PVDF/10 wt% BaTiO<sub>3</sub> with a maximum output voltage of 50 V during repetitive bending and releasing at 4 Hz and about 4 V under the frequency of 2 Hz and the pressure of 0.2 N.<sup>55</sup> It must also be pointed out that KNN, which has the same perovskite structure as BaTiO<sub>3</sub>, has also attracted considerable attention. KNN nanorods (NRs) were created through a hydrothermal process and used to create electrospun PVDF/KNN nanofiber films. It was discovered that the maximal output voltage was approximately 17.5 V at an applied force of 0.5 N and a frequency of 10 Hz for the PVDF/3 wt% KNN NRs films with the  $\beta$ -phase content of 78%.<sup>61</sup>

In lead-free piezoelectric ceramics, ZnO with a good piezoelectric coefficient ( $d_{33} = 12.4 \text{ pm V}^{-1}$ ) also exhibits great application prospects. It has been used in sunscreens with FDA

approval, so it is safe for people.<sup>62–65</sup> The ZnO nanoparticles (NPs) can be added to PVDF to improve the piezoelectric output performance (Fig. 2a). It was found that ZnO NPs can operate as nucleating agents to produce strong interactions on the contact surface with PVDF, and increase the  $\beta$ -phase content of PVDF.<sup>66</sup> The peak output voltage and current of the PVDF/ZnO device were 4.4 V and 45  $\mu\text{A}$ , while those of the pure PVDF device were only 1.7 V and 160 nA at a force of 5 N and a frequency of 0.4 Hz. In addition to the addition of ZnO NPs, ZnO nanowires (NWs) could also be added into electrospun PVDF and nylon 11 nanofibers to develop innovative piezoelectric nanofiber materials. Both PVDF/ZnO NWs and nylon/ZnO NWs were processed into triboelectric nanogenerators (TENGs).<sup>67</sup> The addition of ZnO NWs made the  $\beta$ -phase content increase from 61.3% to 73.4%. The maximum open-circuit voltage of the nanogenerator was 330 V. ZnO NRs could also be added as a useful filler to enhance the piezoelectric capabilities of PVDF fiber films with an open-circuit voltage of roughly 85 V and a short-circuit current of roughly 2.2  $\mu\text{A}$ . The  $\beta$ -phase content of PVDF/ZnO NRs fibrous films could reach 90.7%.<sup>68</sup>

Currently, organic piezoelectric fillers are also used in electrospun PVDF nanofiber films. There was a study which reported the preparation of PVDF/PAN composite nanofiber films through electrospinning. The results showed that the  $\beta$ -phase content, fracture strength, and elongation of the film were 83.4%, 7 MPa and 26%, respectively, which were dominant and could be used to prepare PENGs.<sup>69</sup> When the applied force was 1 N, the PENG had a stable output voltage of 1.3 V. Cellulose nanocrystals (CNCs) as an organic piezoelectric nanofiller can also be used to prepare PVDF/CNC composite fibers. Researchers explored the influence of doping different concentrations of CNCs on the piezoelectric output performance of PVDF fiber films. It was found that when the CNC concentration was 2 wt%, the highest  $\beta$ -phase content of 89.96% could be obtained with the output voltage of 60 V.<sup>70</sup>



Fig. 2 Electrospun PVDF-based single-filler nanofiber films: (a) schematic of the proposed mechanism for the electrospinning of pure PVDF, ZP30/PVDF, ZR5/PVDF composite fibers. Reproduced from ref. 68 with permission from the American Chemical Society, copyright 2019. (b) Three-dimensional molecular model of the  $\beta$ -phase formation during the electrospinning process. Reproduced from ref. 77 with permission from the Elsevier Ltd, copyright 2020. (c) Interaction mechanism between talc nanosheets and PVDF chain in electrospinning nanocomposite fabric. Reproduced from ref. 87 with permission from the Royal Society of Chemistry, copyright 2020.



**2.1.2 Non-piezoelectric materials.** Inorganic and organic substances are the two categories of non-piezoelectric doping materials. Specifically, inorganic materials include carbon materials, two-dimensional MXene materials, salt compounds containing metal ions, clay materials, *etc.* Organic materials include polyurethane, conductive polymers, *etc.*

Because graphene and carbon nanotubes (CNTs) are carbon materials with excellent electrical conductivity, the  $\beta$ -phase content and the charge transfer capability were found to be simultaneously increased for PVDF nanofiber films doped with them.<sup>71,72</sup> It was shown that multi-walled CNTs (MWCNTs) could act as a nucleation in the crystallization process facilitating the conversion of the  $\alpha$  phase to the  $\beta$  phase and cause charge accumulation at the interface when they were used.<sup>37</sup> Therefore, high  $\beta$ -phase content of PVDF films could be acquired by adding MWCNTs. It was found that the  $\beta$ -phase content of the PVDF/5 wt% MWCNT nanofiber film was 68.4% and could produce a maximum piezoelectric voltage of up to 6 V at a mechanical bending frequency of 0.8 Hz. However, the  $\beta$ -phase content and piezoelectric voltage of the pure PVDF nanofiber film were only 54.8% and 2 V.<sup>73</sup> There has been only limited study on CNTs as a single-filler doping in recent years; extensive studies are focusing on graphene as a candidate. Researchers prepared graphene oxide (GO) and reduced graphene oxide (rGO) materials through Hummers' method and then made them into PVDF/GO and PVDF/rGO composite nanofibers by electrospinning.<sup>74</sup> According to the findings, the open-circuit voltage was found to increase from 0.50 V to 0.65 V and 4.38 V by respectively adding 0.8 wt% GO and rGO to PVDF at a finger tap of 5 Hz. Another researcher increased the content of GO or rGO in the PVDF fiber film to 2 wt%, and the  $\beta$ -phase contents of PVDF/GO and PVDF/rGO were both increased to about 87%.<sup>75</sup> The short-circuit current and open-circuit voltage of the PVDF/GO fiber film were 200 nA and 1.5 V, respectively. Due to the higher conductivity of rGO, the short-circuit current and open-circuit voltage of the PVDF/rGO fiber film were 700 nA and 16 V, respectively, which were 20 times higher than those for the pure PVDF film. Therefore, rGO has a better impact on the piezoelectric output augmentation of PVDF fiber films.

MXene, rich in a large number of  $-\text{OH}$  groups can yield hydrogen bonding interactions with  $-\text{CH}_2-$  in PVDF chains and facilitate the formation of the  $\beta$  phase in PVDF (Fig. 2b).<sup>76</sup> The oriented PVDF-TrFE/MXene fiber film was prepared by electrospinning, and the results showed that when the addition of MXene was 2 wt%, the output voltage of the PVDF-TrFE/MXene composite film was 1.58 V at a frequency of 1 Hz and a force of 20 N, which was 3 times that of the pure PVDF-TrFE fiber film under the same mechanical stimulation.<sup>77</sup>

The  $\beta$ -phase content of PVDF nanofiber films could also be increased by mixing salt compounds with metal ions into the PVDF matrix.<sup>78–80</sup> It was found that the piezoelectric qualities of PVDF fiber films could be substantially boosted by doping silver nitrate ( $\text{AgNO}_3$ ).<sup>81</sup> Under a shaker power unit with a frequency of 500 Hz and an amplitude of 5 mm, the output voltage was increased from 0.27 V to 2 V by adding 0.3 wt%  $\text{AgNO}_3$ . Adding  $\text{BiCl}_3$  to PVDF nanofiber composite films could also enhance

their  $\beta$ -phase content. The PVDF/ $\text{BiCl}_3$  composite film PENG obtained an output voltage of 1.1 V under vertical vibration conditions, which was 4.76 times higher than that of the pure PVDF film PENG.<sup>82</sup> According to a study, lithium chloride ( $\text{LiCl}$ ) could also contribute to the increase of the amount of  $\beta$  phase for the electrospun PVDF nanofibers.<sup>83</sup> The  $\beta$ -phase content of PVDF nanofiber films increased from 84% to 93.6% when  $\text{LiCl}$  was added at 0.00133 wt%. The piezoelectric output signal of the fiber films was tested by selecting 12 g spheres dropped from a height of 30 cm, and the results showed that the peak output voltages of PVDF/ $\text{LiCl}$  and PVDF fiber films were 8 V and 1.89 V, respectively.

In addition, clay materials were frequently used as fillers.<sup>84–86</sup>  $\text{Mg}_3\text{Si}_4\text{O}_{10}(\text{OH})_2$  (Talc), a clay material, is rich in  $-\text{OH}$  group and can interact with  $-\text{CF}_2$  group of the PVDF matrix through hydrogen bonds, thus contributing to the orientation of dipoles in the PVDF matrix and the formation of the  $\beta$  phase (Fig. 2c). The  $\beta$ -phase content of the PVDF composite nanofiber film with 0.5 wt% Talc reached 89.6%, higher than that of pure PVDF (60.4%). The maximum output voltage of the composite film was 9.1 V, whereas the output voltage of pure PVDF was only 1.6 V at a finger tap force of about 3.8 N.<sup>87</sup>

Organic polymers and their derivatives with  $\text{C}=\text{O}$  or  $-\text{NH}_2$  groups can interact with  $-\text{CH}_2$  and  $-\text{CF}_2$  groups in PVDF to form hydrogen bonds, thus increasing the  $\beta$ -phase content.<sup>88</sup> It was found that the  $\beta$  phase of electrospun PVDF/TPU nanofibers was formed during the chemical interaction between the  $-\text{CF}_2$  group of PVDF and the  $-\text{NH}_2$  group in TPU.<sup>89</sup> Furthermore, intrinsically conducting polymers (ICPs) could be spun as a filler with PVDF to increase the efficiency of charge transfer and the piezoelectric responsiveness of PVDF.<sup>90,91</sup> Polycarbazole (PCZ) and polyaniline (PANI) were added respectively to prepare electrospun PVDF/PCZ and PVDF/PANI fiber films. Studies have shown that the PVDF/PCZ fiber film had a higher  $\beta$ -phase content than PVDF/PANI and pure PVDF fiber films.<sup>92</sup> The open-circuit output voltage of PVDF/PCZ fiber film was 2 V under the parameters of effective area of  $2.7 \text{ cm} \times 2.3 \text{ cm}$ , pressure of 0.24 MPa, and frequency of 4 Hz, but that of the pure PVDF nanofiber was only 0.35 V.

For the single-filler doping, some researchers have chosen fillers that can act as nucleating agents to promote the enhancement of  $\beta$ -phase content in PVDF fiber films, while others have selected fillers with excellent electrical conductivity to promote charge transfer. For better results, researchers have explored the means of double-filler doping to play the synergistic effect of the above two aspects.

## 2.2 Double-filler doping

There are a variety of double-filler combination methods, and each combination method has a unique set of benefits compared with single-filler doping. The first is to use two piezoelectric materials to enhance piezoelectric capabilities. The second is to utilize conductive fillers to more quickly and effectively transfer the charge from the inside of the fiber film to its surface leading to an increase in output voltage. The third is to use other materials to modify the piezoelectric materials and



play the role of interface modification or ion binding to improve the piezoelectric properties.

**2.2.1 Dual piezoelectric materials.** Dual piezoelectric material filling is a combination method that some researchers have tried to use to improve the piezoelectric output. As mentioned above, a single piezoelectric filler has an enhancing effect on the piezoelectric performance of PVDF fiber films. Combining two proven effective piezoelectric materials for co-doping can have a synergistic effect in enhancing the piezoelectric output of PVDF fiber films.

The addition of BaTiO<sub>3</sub> and ZnO NRs in PVDF solution can facilitate the linking of polymer molecular chains to the crystalline  $\beta$ -phase due to contact between PVDF, BaTiO<sub>3</sub>, and ZnO NRs to improve the formation of the  $\beta$  phase in electrospun nanofibers.<sup>93</sup> The results demonstrated that the piezoelectric output of the PVDF/BaTiO<sub>3</sub>/ZnO composite film increased to 12 V at an applied force of 1.5 N. In another study on the electrospun PVDF/KNN/ZnO composite film, its output voltage was found to be higher than that of PVDF/KNN or PVDF/ZnO (Fig. 3a).<sup>94</sup> The PVDF/3% KNN/2% ZnO fiber film had the highest  $\beta$ -phase content of 94% and the output voltage and current of the nanogenerator made of it were 25 V and 1.81  $\mu$ A in a finger tapping mode and 8.31 V and 5  $\mu$ A at a constant pressure of 1 kPa. Consequently, the piezoelectric capabilities of PVDF/KNN/ZnO nanogenerators were significantly improved by the synergistic impact of KNN and ZnO NRs in the PVDF polymer matrix.

**2.2.2 Piezoelectric materials and conductive materials.** Piezoelectric materials and conductive materials can be simultaneously doped into the electrospun PVDF nanofibers. The conductive materials can promote the formation of the  $\beta$  phase and improve electron transport capacity leading to a significant increase of the output voltage.<sup>95</sup>

The addition of CNTs as a conductive material into the PVDF/KNN spinning solution enhanced the conductivity of the polymer jet, leading to a higher  $\beta$ -phase content in the hybrid nanocomposite (Fig. 3b).<sup>96</sup> The PVDF/KNN fiber nanogenerator produced a voltage of about 17.5 V and a current of about 0.522  $\mu$ A by doping 3 wt% KNN NRs in PVDF. When 0.1 wt% CNTs were further doped into PVDF/KNN solution to spin PVDF/KNN/CNT composite nanofiber films, the output voltage could be up to about 23.24 V and the current was about 9  $\mu$ A when tapped by a finger.

Graphene with an excellent electrical conductivity has also been used by combining with piezoelectric materials to prepare composite nanofiber films. Graphene-ZnO (G-ZnO) nanocomposites were synthesized by a hydrothermal method and then added to the PVDF nanofibers through electrospinning. The piezoelectric capability of PVDF/G-ZnO nanofiber composites was improved due to the higher  $\beta$ -phase content (62.36%) compared with that of PVDF/ZnO (56.22%) and pure PVDF (51.08%).<sup>97</sup> The PVDF/G-ZnO nanofiber composites had an average output voltage of 840 mV under an external force of 1 N, which was superior to PVDF/ZnO (710 mV) and pure PVDF (640 mV) nanofibers. Graphene nanosheets and BaTiO<sub>3</sub> NPs can also be co-incorporated into PVDF to prepare PVDF/graphene/BaTiO<sub>3</sub> composite nanofiber films. When the contents of graphene and BaTiO<sub>3</sub> were 15 wt% and 0.15 wt%, the  $\beta$ -phase content of PVDF/graphene/BaTiO<sub>3</sub> composite nanofiber film could reach 91.1%. The open-circuit voltage and power were respectively up to 11 V and 4.1  $\mu$ W at a loading frequency of 2 Hz and a deformation of 4 mm. Under 1800 cycles in the durability test, there was no appreciable drop of open-circuit voltage. Furthermore, the PENG made from the composite film produced a peak voltage of up to 112 V when pressed by a finger, which was enough to power an electric meter or illuminate 15

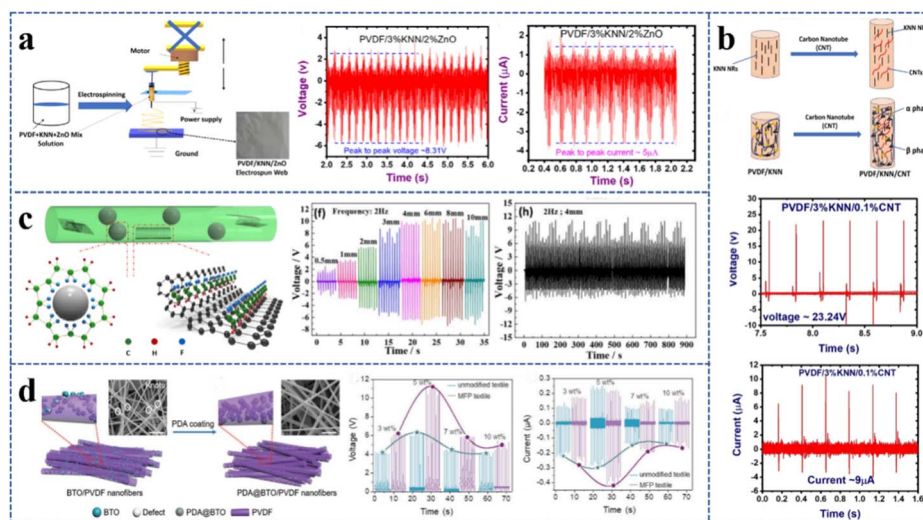


Fig. 3 Electrospun PVDF-based nanofiber films with two fillers: (a) fabrication of electrospun PVDF/KNN/ZnO web. Reproduced from ref. 94 with permission from John Wiley & Sons, Inc., copyright 2020. (b) Mechanical stretching effect of the electrospun fiber in the presence of CNTs and  $\beta$ -nucleating effect of the CNTs in the PVDF polymer matrix. Reproduced from ref. 96 with permission from the Royal Society of Chemistry, copyright 2020. (c) The mechanism diagram of  $\beta$  phase formation on BaTiO<sub>3</sub> NPs and graphene nanosheets in the nanocomposite fiber. Reproduced from ref. 98 with permission from Elsevier Ltd, copyright 2018. (d) Comparison between unmodified and modified electrospun fibers via dopamine coating. Scale bar is 1  $\mu$ m. Reproduced from ref. 101 with permission from John Wiley & Sons, Inc., copyright 2021.



LEDs. Therefore, the combination of graphene nanosheets with BaTiO<sub>3</sub> NPs was demonstrated to enhance the piezoelectric capabilities of composite nanofibers (Fig. 3c).<sup>98</sup>

**2.2.3 Piezoelectric materials and other materials.** In addition to the two combination methods mentioned above, researchers have also combined piezoelectric materials with other materials (such as organic compounds and salt compounds with metal ions) to improve the piezoelectric output capacity of PVDF fiber films.<sup>99,100</sup>

Inspired by biological connective tissue, a thin interfacial connection based on polydopamine (PDA) was created between BaTiO<sub>3</sub> NPs and the PVDF matrix.<sup>101</sup> The PDA coating was not only conducive to the improvement of the bonding force of BaTiO<sub>3</sub> and PVDF but also could enable the composite fiber to obtain a favorable smooth surface morphology (Fig. 3d), which was advantageous to the load transfer in the PDA@BaTiO<sub>3</sub>/PVDF composite fiber and the generation of a stronger electromechanical coupling effect, thus obtaining a higher piezoelectric response. The findings demonstrated that the piezoelectric performance increased by 47% when 3.02 wt% PDA was doped. The output voltage of the PDA-modified device was around 11 V, whereas that of the unmodified device was around 6.5 V when the mass fraction of BaTiO<sub>3</sub> was 5 wt%.

PVDF, PVDF/ZnO, PVDF/BiCl<sub>3</sub>, and PVDF/BiCl<sub>3</sub>/ZnO nanofibers were separately spun in order to study the impact of ZnO and BiCl<sub>3</sub> on the piezoelectric capabilities of PVDF nanofiber films.<sup>102</sup> The findings showed that the  $\beta$ -phase content of PVDF/BiCl<sub>3</sub>/ZnO nanofiber films was up to 92%, and those of pure PVDF, PVDF/BiCl<sub>3</sub>, and PVDF/ZnO were 76%, 84%, and 88%, respectively. The active interaction between the surface charges of ZnO NPs and the ionized ions produced by BiCl<sub>3</sub> was responsible for the high  $\beta$ -phase content of PVDF/BiCl<sub>3</sub>/ZnO nanofiber films. The open-circuit voltage and short-circuit

current of the PVDF/BiCl<sub>3</sub>/ZnO nanofiber films were around 12 V and 80 nA, which were obviously higher than the values of 4 V and 20 nA for pure PVDF ones.

As mentioned above, researchers have attempted to improve the piezoelectric performance of PVDF fiber films by double-filler doping, aiming at a breakthrough in the functionality of PVDF-based piezoelectric devices just with a single filler. It is clear from the literature that researchers are paying more attention to the combination of piezoelectric and conductive materials, or piezoelectric and other materials. However, investigations on dual piezoelectric fillers have not been carried out in great numbers.

The comparison of  $\beta$ -phase contents and output voltages of some electrospun filler-modified PVDF nanofiber films mentioned above under different pressure conditions is shown in Table 1.

### 3. Structural modification of PVDF nanofiber films

The output performance of the electrospun PVDF piezoelectric nanofiber film can also be improved by modifying its structures. Nanofibers with a highly oriented structure can be obtained by changing the receiving device of the electrospinning setups. Coaxial nanofibers can be obtained by changing the needle form of electrospinning. Multilayer nanofibers can be produced by changing the spinning method or combining it with other techniques.

#### 3.1 Highly oriented structure

By changing receiving devices, the microscopic morphology of electrospun PVDF fiber films can be altered. Disordered

**Table 1** Comparison of  $\beta$ -phase contents and output voltages of PVDF-based nanofiber films through filling modification

Filling types	Nanofiber films	$\beta$ -Phase contents (%)	Pressure conditions	Output voltages (V)	References	
Single filler	PVDF/PZT	—	2.125 N	0.184	38	
	PVDF/BaTiO <sub>3</sub>	91.0	0.2 N 2 Hz	~4	60	
	PVDF/KNN	78	0.5 N 10 Hz	~17	61	
	PVDF/ZnO	80.2	5 N 0.4 Hz	4.4	66	
	PVDF/PAN	83.4	1 N	1.3	69	
	PVDF/CNCs	89.96	—	60	70	
	PVDF/CNTs	68.4	0.8 Hz	6	73	
	PVDF/GO	87	—	1.5	75	
	PVDF/rGO	87	—	16	75	
	PVDF-TrFE/MXene	—	20 N 1 Hz	1.58	77	
	PVDF/AgNO <sub>3</sub>	—	500 Hz 5 mm	2.0	81	
	PVDF/BiCl <sub>3</sub>	85.7	—	1.1	82	
	PVDF/LiCl	93.6	12 g 30 cm	8	83	
	PVDF/Talc	89.6	~3.8 N	9.1	87	
	PVDF/PCZ	—	0.24 MPa 4 Hz	2	92	
	Double fillers	PVDF/ZnO/BaTiO <sub>3</sub>	—	1.5 N	12	93
		PVDF/KNN/ZnO	94	1 kPa	8.31	94
PVDF/KNN/CNTs		84	1 kPa	12	96	
PVDF/G-ZnO		62.36	1 N	0.84	97	
PVDF/BaTiO <sub>3</sub> /graphene		91.1	2 Hz	11	98	
PDA@ BaTiO <sub>3</sub> /PVDF		63.0	3 N	~13	101	
PVDF/BiCl <sub>3</sub> /ZnO		92	0.2 N	12	102	



nanofibers are usually collected by using flat-plate receiving devices, while oriented nanofibers can be collected by using high-speed drums, spinning discs, Helmholtz coils, twin rod collectors, and stepped collecting devices. The output performance of PVDF fiber films is proven to be affected by the orientation degree of nanofibers. The higher the orientation degree of the fibers, the more rapid the charge transfer in the axial direction of the fibers and the higher the piezoelectric output.

One of the most popular collectors for gathering oriented electrospun fibers is the high-speed rotating drum.<sup>103</sup> It was discovered that the output voltage of oriented PVDF fiber films obtained at a drum speed of 1500 rpm rose from 0.8 V to 3.6 V as the acceleration amplitude of the shaker increased from 2.0 g to 3.0 g (Fig. 4a).<sup>104</sup> When the drum speed was increased to 4000 rpm, highly oriented PVDF-TrFE nanofiber films could be formed. The FTIR results showed that the  $\beta$ -phase content of the fiber film was close to 85%. The prepared sensor had good linearity in the low-pressure region (0.1–1 Pa) with a voltage response of 1 V at 1 Hz under a cyclic bending test.<sup>21</sup> Another study further tested the charge output capacity of the oriented nanofiber film by applying two tensile forces (0° parallel and 90° perpendicular to the oriented fiber).<sup>105</sup> The charge output in the 0° direction was 10 pC, higher than that in the 90° direction, demonstrating improved directional strain sensing capabilities for the oriented PVDF-TrFE nanofiber mats (Fig. 4b). The piezoelectric charge output of the nanofiber film along the fiber orientation direction parallel to the cantilever beam was approximately 5 times higher than that along the vertical directions when two PVDF-TrFE nanofiber films of the same size were embedded in the vibrated PDMS cantilever beam, which further indicated that well-aligned PVDF-TrFE nanofiber films had outstanding directional strain sensing capabilities

and the strain sensors made of them exhibited better strain sensitivity than those based on randomly oriented nanofibers.

Oriented nanofibers have also been obtained in other ways. For example, a two-bar system was used to collect oriented PVDF/CNT nanofibers.<sup>106</sup> The results revealed that the  $\beta$ -phase content of random PVDF/CNT nanofibers was 74.2% when 0.3 wt% CNTs were added, but that of the highly oriented PVDF/CNT nanofibers reached 86%. When a pressure of up to 100 g by weight was applied, the sensitivity of random PVDF/CNT nanofibers was 8.8 mV g<sup>-1</sup>, whereas that of strongly oriented PVDF/CNT nanofibers was enhanced to 73.8 mV g<sup>-1</sup> (Fig. 4c).

### 3.2 Core-shell structures

At the fiber level, core-shell structured PVDF nanofibers could be fabricated *via* coaxial electrospinning. In coaxial electrospinning, the spinneret consists of two concentric aligned nozzles, and the core and sheath precursor solutions are respectively fed through the inner nozzle and the outer nozzle at a controlled rate. This structure has been reported to contribute to the increase in the output voltage of fibers.

One way the researchers designed the core-shell structure was to introduce nano fillers in the core-shell PVDF spinning solution. BaTiO<sub>3</sub> NPs and GO nanosheets were jointly introduced in PVDF, with PVDF/BaTiO<sub>3</sub> and PVDF/GO solution as the core and skin layers, respectively. The resultant core-shell structured nanofibers were found to have greatly improved piezoelectric characteristics (Fig. 5a). In comparison to pure PVDF nanofibers, the output voltage of the core-shell structured nanofibers increased from 0.75 V to 2.5 V and the current increased from 4.6 nA to 10.5 nA at a frequency of 2 Hz and an applied external force of 10 N.<sup>107</sup> In another study, (PVDF-HFP)-TiO<sub>2</sub>/PVDF-ZnO core-shell nanofibers were prepared by adding TiO<sub>2</sub> and ZnO NPs to PVDF-HFP and PVDF, respectively.<sup>108</sup> The

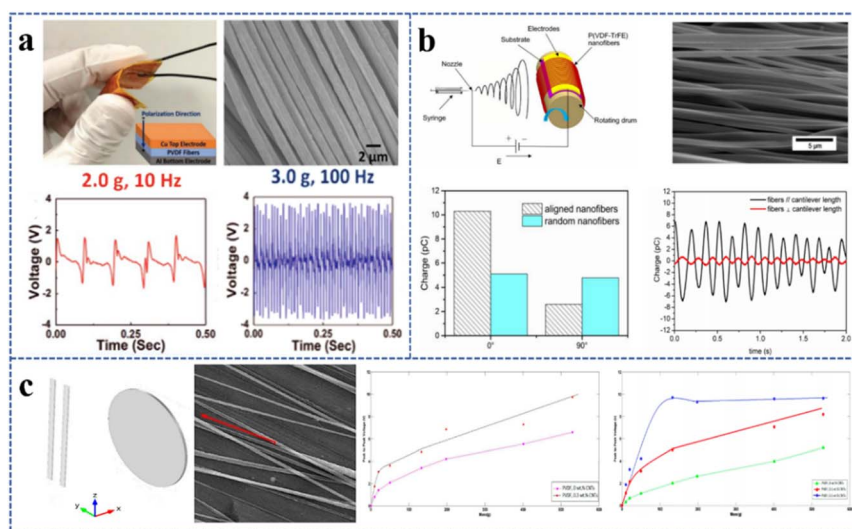


Fig. 4 Electrospun highly oriented PVDF-based nanofiber films: (a) piezoelectric properties of electrospun oriented PVDF fiber film. Reproduced from ref. 104 with permission from John Wiley & Sons, Inc., copyright 2020. (b) Aligned PVDF-TrFE nanofibers for enhanced piezoelectric directional strain sensing. Reproduced from ref. 105 with permission from the MDPI, copyright 2018. (c) Aligned electrospun PVDF/CNT nanofibers collected by a two-bar system. Reproduced from ref. 106 with permission from the MDPI, copyright 2018.





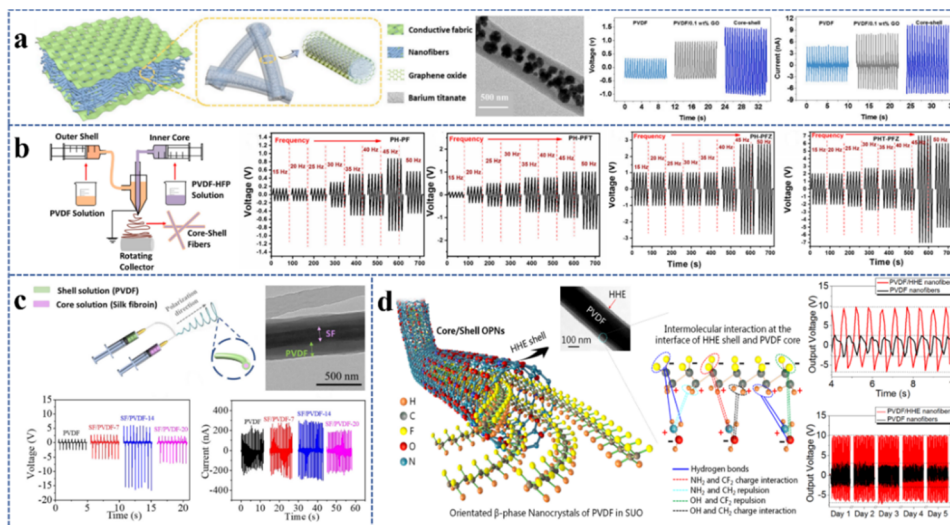


Fig. 5 Electrospun core-shell structured PVDF-based nanofiber films: (a) PVDF-BaTiO<sub>3</sub>/PVDF-GO piezoelectric nanofiber film. Reproduced from ref. 107 with permission from Elsevier Ltd, copyright 2020. (b) Piezoelectric properties of (PVDF-HFP)-TiO<sub>2</sub>/PVDF-ZnO nanofiber film. Reproduced from ref. 108 with permission from the MDPI, copyright 2020. (c) PVDF/SF coaxial piezoelectric nanofiber film. Reproduced from ref. 109 with permission from Chongqing University, copyright 2021. (d) PVDF/HHE coaxial piezoelectric nanofiber film. Reproduced from ref. 110 with permission from the American Chemical Society, copyright 2019.

maximum output voltage of PVDF-HFP/PVDF core-shell nanofibers without fillers was 1.75 V at 45 Hz, while that of the (PVDF-HFP)-TiO<sub>2</sub>/PVDF core-shell nanofibers was slightly increased to 2 V, the (PVDF-HFP)/PVDF-ZnO core-shell nanofibers rose to 5.5 V, and the output voltage of the composite film material containing both TiO<sub>2</sub> and ZnO reached 14 V (Fig. 5b).

Another way is that some kinds of other polymer solutions are combined with PVDF solution to form a fiber with a core-shell structure. The formation of  $\beta$  phase in the core-shell structure was facilitated by the intermolecular contact between the shell polymer layer and the core PVDF layer, hence increasing the content of  $\beta$  phase in PVDF. Core-shell structured piezoelectric nanofibers were obtained by coaxial electrospinning of SF and PVDF solutions as shown in Fig. 5c.<sup>109</sup> When the SF content was 14 wt%, the  $\beta$ -phase content reached 66%. Compared with the output voltage and current of 2.5 V and 190 nA for pure PVDF nanofiber devices, those for PVDF/SF nanofiber devices were 16.5 V and 290 nA showing a significant increase. This nanofiber film was also quite flexible. Another study improved the stability of the  $\beta$  phase in PVDF by wrapping hydroxylamine hydrochloride (HHE) over the outer surface of PVDF nanofibers to form a thin shell.<sup>110</sup> The molecular interactions of the -NH<sub>2</sub> group in HHE and the -CF<sub>2</sub> group in PVDF increased the  $\beta$ -phase content of PVDF/HHE piezoelectric nanofibers to 96% at the HHE concentration of 1.0 wt%, and the output voltage measured at a micropressure of 1 kPa and a low frequency of 1.5 Hz was 18 V, which was 3.6 times that of the pure PVDF nanofiber film (Fig. 5d). The PVDF/HHE sensor demonstrated outstanding wear resistance throughout 5 days of nonstop cyclic compression (28 000 cycles), with the output voltage remaining nearly constant at 18 V, but that of the pure PVDF sensor gradually declined from 5.1 V to 3.5 V.

### 3.3 Multilayer composite structure

Utilizing the top outside layers of the multilayer composite fiber film as electrode materials is one purpose for the design of a multilayer composite structure. Another aim is to increase the total dipole moment for a higher piezoelectric output by increasing the number of layers of multilayer composite fiber films.

By combining the electrospun PVDF fiber film with the fiber film generated in another way (*e.g.*, vapor phase polymerization, electrospinning, and tape casting), a multilayer composite structure can be created. It was found that electrospinning and vapor phase polymerization could be combined to improve the output performance of all-organic piezoelectric nanogenerators (OPNG). Vapor phase polymerization was utilized to deposit poly(3,4-ethylenedioxythiophene) (PEDOT) on the surface of electrospun PVDF nanofibers to prepare PEDOT-coated PVDF as the electrode and auxiliary piezoelectric layer. A layered structure of PEDOT-coated PVDF/PVDF/PEDOT-coated PVDF was created, with pure PVDF nanofibers sandwiched in the middle as the primary piezoelectric layer (Fig. 6a). The OPNG displayed an output voltage of roughly 48 V when a pressure of 8.3 kPa was applied. Over 21 000 cycles of continuous mechanical shock were used in the 6-month-long fatigue testing, which revealed a steady piezoelectric output and a good durability.<sup>111</sup> In addition, electrospinning, vapor phase polymerization, and electrospinning techniques could be combined to prepare a multilayer structure. Researchers created a high-performance nanogenerator made of an electrospun PVDF nanofiber layer in the middle, polypyrrole (PPy)/PVDF nanofibers through vapor phase polymerization of the pyrrole monomer on the surface of electrospun PVDF nanofibers as the top/bottom electrodes, and two layers of electrospun bromine lead cesium(CsPbBr<sub>3</sub>)



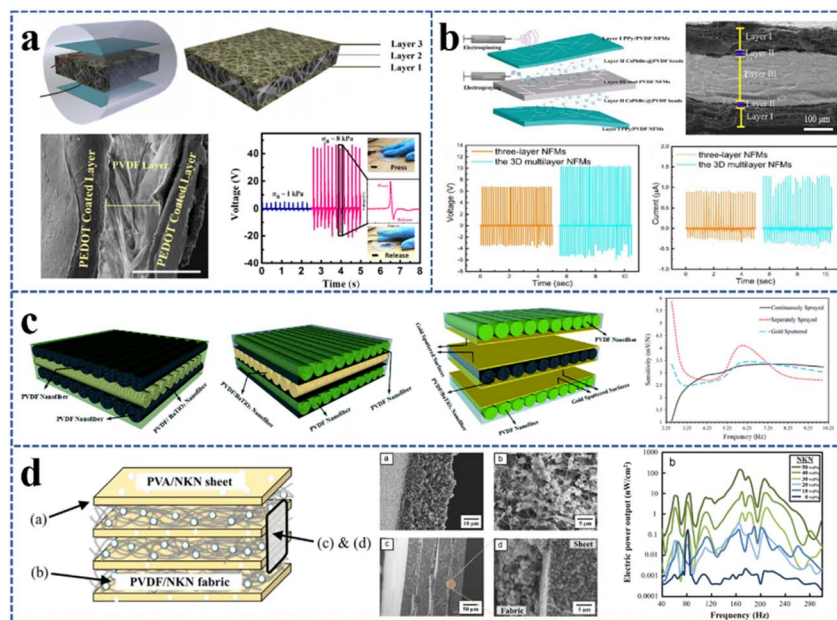


Fig. 6 Electrospun PVDF-based nanofiber films with a multilayer structure: (a) multilayer assembled electrospun nanofiber mats and their piezoelectric properties. Reproduced from ref. 111 with permission from the American Chemical Society, copyright 2018. (b) Multilayer assembly of electrospun/electrosprayed PVDF-based nanofibers and beads. Reproduced from ref. 112 with permission from Elsevier B.V., copyright 2020. (c) Flexible PVDF/BaTiO<sub>3</sub> hybrid-structure pressure sensor. Reproduced from ref. 113 with permission from the Royal Society of Chemistry, copyright 2020. (d) PVDF-based multilayer composite. Reproduced from ref. 114 with permission from Elsevier B.V., copyright 2015.

@PVDF microbeads inserted between them (Fig. 6b).<sup>112</sup> According to the piezoelectric test results, the nanogenerator was found to be capable to detect weak pressures as low as 7.4 Pa and had a high open-circuit voltage of 10.3 V and a short-circuit current density of  $1.29 \mu\text{A cm}^{-2}$ .

The multilayer composite structure can also be formed by the superposition of electrospun PVDF fiber films doped with different fillers or the superposition of electrospun PVDF fiber films with other electrospun fiber films. For instance, piezoelectric sensors were fabricated from three types of triple-layer electrospun PVDF/PVDF-BaTiO<sub>3</sub>/PVDF nanofiber composites, which were created by sandwiching two PVDF fiber layers with a PVDF/BaTiO<sub>3</sub> fiber layer (Fig. 6c)<sup>113</sup> in which, the second combination had the best varistor performance with a sensitivity of  $6 \text{ mV N}^{-1}$ . Besides, electrospun PVDF fiber films can be stacked with other electrospun fiber films to improve the piezoelectric output. By alternatively laminating electrospun PVDF/NKN fiber films with PVA/NKN sheets created by tape casting, researchers have created multilayer composites for energy harvesting (Fig. 6d). The greatest power output of the energy harvester was measured to be  $145 \text{ nW cm}^{-2}$  in a vibration cycle of 170 Hz.<sup>114</sup>

Based on the above research studies, it can be seen that the output performance of the PVDF piezoelectric nanofiber film can be improved through structural design. PVDF fiber films with a highly oriented structure, coaxial structure or multilayer composite structure have higher piezoelectric output than those with a disordered structure. Moreover, some doped fillers can combine with the structure to play a synergistic role in the performance of PVDF piezoelectric fiber films.

## 4. Applications

### 4.1 Health care

PVDF nanofiber films can be used to make biocompatible and flexible implantable sensors to be implanted in animals to detect minute internal pressures and respond to the beating and contraction of the heart and blood vessels for pathological detection,<sup>115–117</sup> or to promote wound healing using the generated electrical stimulation signals.<sup>118,119</sup> Similarly, external detection sensors prepared from PVDF fiber films can be used for health monitoring of pressure changes generated by human movement in daily life, such as breathing, foot pressure, and flexion of fingers, elbow, and neck.<sup>120</sup> Compared with sensors for monitoring human motion, implantable sensors and sensors for detecting tiny pressure such as pulse need higher sensitivity to achieve detection function.

Regarding implantable sensors, the PVDF/HHE sensor implanted in experimental pigs maintained its hypersensitivity, allowing real-time and accurate detection of internal micro-pressure changes outside the cardiovascular wall (Fig. 7a), as well as simultaneous output of piezoelectric signals reflecting changes in cardiovascular elasticity, cardiac impulse, and thrombosis.<sup>110</sup> The researchers subsequently implanted a flexible PVDF/DA piezoelectric sensor into mice and achieved exceptional sensitivity and accuracy in detecting weak physiological mechanical impulses brought on by diaphragmatic motion and blood pulsation, which were crucial for early diagnosis and medical prevention of acute and chronic cardiovascular diseases.<sup>8</sup> Furthermore, the mechanical stability and biocompatibility of the PVDF/DA sensor were confirmed. The





Fig. 7 Application of electrospun PVDF-based nanofiber films in medical and health fields: (a) PVDF/HHE sensor for real-time micropressure monitoring of cardiovascular walls. Reproduced from ref. 110 with permission from the American Chemical Society, copyright 2019. (b) Wearable sensor prepared from the PVDF/BiCl<sub>3</sub>/ZnO nanofiber film used for the detection of the bending and stretching of the left arm and left leg. Reproduced from ref. 102 with permission from Elsevier Ltd, copyright 2021. (c) PVDF-HFP/ZnO composite nanofiber-based highly sensitive piezoelectric sensor for wireless workout monitoring. Reproduced from ref. 122 with permission from the Springer Nature Switzerland AG, copyright 2021. (d) PVDF/PDA@BaTiO<sub>3</sub> wearable pressure sensor used to monitor physiological signals. Reproduced from ref. 101 with permission from the Elsevier John Wiley & Sons, Inc., copyright 2021. (e) MMG sensors prepared using electrospun PVDF piezoelectric nanofibers for lower limb rehabilitation exoskeleton. Reproduced from ref. 123 with permission from Elsevier B.V., copyright 2019.

immunological reaction was relatively mild six weeks after the device was implanted, and it could continue to work as intended.

Bioactive piezoelectric materials with charged surfaces can produce electrical charges under mechanical strain. When they are implanted in animals, the electrical stimulation signal will encourage cell division, proliferation, and regeneration processes, which can speed up the healing process of wounds. Electrospun PVDF/PU scaffolds have been demonstrated to be effective for wound healing. Animal movement caused mechanical deformation in the scaffold and generated piezoelectric stimulation, which could enhance cell migration and promote the activity of fibroblasts leading to wound healing.<sup>121</sup>

The sensors made of electrospun PVDF fiber films can also be used for *in vitro* monitoring of various physiological activities of the human body. PVDF/BiCl<sub>3</sub>/ZnO sensors were placed on elbows and knees to monitor human motion. The elbow bending from 0° to 40° and from 0° to 80° resulted in peak output voltage signals of 1.2 V and 3 V, respectively, and the knee bending from 0° to 80° also produced a peak output voltage of 0.6 V (Fig. 7b).<sup>102</sup> Piezoelectric sensors made of PVDF-HFP/ZnO composite nanofibers had a high sensitivity of 1.9 V kPa<sup>-1</sup>. When subjected to a force ranging from 0.02 to 0.5 N, a quick response time of 20 ms was achieved. It could last up to 5000 times of cycle test, indicating high stability and durability. This high-sensitivity sensor was used to track the movement of athletes in real time to minimize the risk of damage. The results showed that the output signal values in the physically exhausted condition were noticeably lower than those in the normal state (Fig. 7c).<sup>122</sup>

Wearable pressure sensors made of electrospun PVDF/PDA@BaTiO<sub>3</sub> nanofibers had an excellent endurance (less than 3% drop after 7400 cycles) and a great sensitivity (3.95 V N<sup>-1</sup>).<sup>101</sup> The sensors could detect the bending and stretching of the finger with the output voltage from ~1.5 V to ~3 V with the increase of bending angle from 30° to 90°. It was also observed that the vibration of the throat caused the sensor to produce different signal profiles when the sensor was held to the throat and different phrases were spoken (Fig. 7d). Other researchers used electrospun PVDF fibers to prepare a mechanomyography (MMG) sensor that might be as a motion switch trigger sensor for a lower limb rehabilitation exoskeleton (LLRE). Comparing this trigger sensor with a commercial electromyography (EMG) sensor, it was found that the maximum signal amplitude of the commercial EMG sensor was about 0.2 V, while that of the MMG sensor was about 2.8 V (Fig. 7e).<sup>123</sup>

## 4.2 Environmental monitoring

A flexible PVDF sensor was found to produce various piezoelectric signals in response to varying sound, wind, or humidity in the daily environment, and could be used to monitor the environmental conditions.<sup>124</sup> The vibration generated by sound or wind is very weak, so the sensitivity of the corresponding sensor is required to be high. The humidity sensor is mainly influenced by the characteristics of additives in the PVDF polymer matrix, thus the performance of additives plays a key role in determining the quality of the device.

To track the faint vibration of a loudspeaker, a flexible PVDF/BaTiO<sub>3</sub> pressure sensor with a sensitivity of 18.0 V N<sup>-1</sup> was developed.<sup>60</sup> By attaching the sensor to a stereo, the maximum



periodic output voltages of 150 mV and 500 mV could be obtained when the music was played at different sound pressure levels of 60 dB and 80 dB. It was even feasible to discern between different genres of popular music (Fig. 8a). Besides, some researchers prepared a PVDF/graphene/Ce<sup>3+</sup> nanofiber pressure sensor, which had outstanding piezoelectric capabilities with an output voltage of 11 V and a maximum power of 6.8  $\mu$ W at the pressure of 6.6 kPa and achieved an ultra-high pressure measurement sensitivity even at 2 Pa. It could generate different voltage signals under different wind speeds for environmental wind monitoring, and could also identify the same song played by different types of musical instruments (flutes, guitars, and violins) (Fig. 8b).<sup>125</sup> A multifunctional sensor was also developed by doping zinc sulfide (ZnS) NRs into PVDF. It was found that the resonance frequency was  $86 \pm 3$  Hz, and the acoustic sensitivity was around  $3 \text{ V Pa}^{-1}$ . It was suitable for noise detection and could distinguish between acoustic waves in the low-frequency and mid-frequency ranges. In addition, the device could also be used for wind speed monitoring. When the wind speed increased from  $15 \text{ m s}^{-1}$  to  $25 \text{ m s}^{-1}$ , the measured output voltage increased from 1.0 to 4.2 V.<sup>126</sup> Some other researchers also built a PVDF piezoelectric film sensor for monitoring wind speed.<sup>127</sup> The findings demonstrated that the output voltage increased with the increase of wind speed at various frequencies. Also, the error between the wind speed measured by the PVDF sensor and that measured by the standard anemometer was less than 5%, demonstrating the accuracy of the PVDF sensor for wind speed monitoring (Fig. 8c).

Due to the hydrophilic properties of MXene, electrospun PVDF-TrFE/MXene sensors can be utilized as humidity-responsive pressure sensors to measure environmental humidity.<sup>77</sup> At varied humidity levels (41%, 84%, and 100%), the voltage signal produced by the enclosed piezoelectric device remained fairly linear to the pressure. In the actual test, the sensor was placed at the heel inside the shoe, and the effect of foot sweating on the output voltage was tested with the increase of jogging time. Results showed that the longer the jogging time, the more serious the foot sweating, and the lower the output voltage (Fig. 8d). Devices made of PVDF/PI nanofiber films have also demonstrated humidity sensing application capabilities. Compared to the pure PVDF device with an open-circuit voltage of about 800 V, the PVDF/PI device could generate a higher value of 1600 V. When the relative humidity fluctuated between 40% and 80%, the output voltage of the PVDF/PI device decreased with the increasing humidity, allowing it to be employed as humidity sensors (Fig. 8e). The output voltage of the PVDF/PI device was also evaluated for 1000 cycles without a significant decrease at a relative humidity of 45%.<sup>128</sup>

### 4.3 Energy harvesting

The market for smart wearables has recently taken off, and experts have been paying close attention to the energy demand of these products. In order to overcome the power supply issue for the majority of smart wearable technology, flexible and sustainable energy harvesting devices should be developed. In



**Fig. 8** Application of electrospun PVDF-based nanofiber films in the environmental monitoring field: (a) flexible PVDF/BaTiO<sub>3</sub> pressure sensor used to monitor the weak vibration of the speaker. Reproduced from ref. 60 with permission from the American Chemical Society, copyright 2020. (b) PVDF/graphene composite nanofibers for fabrication of nanopressure sensors and ultrasensitive acoustic nanogenerators. Reproduced from ref. 125 with permission from the American Chemical Society, copyright 2016. (c) A flow velocity measurement method based on a PVDF piezoelectric sensor. Reproduced from ref. 127 with permission from the MDPI, copyright 2019. (d) PVDF-TrFE/MXene sensor used to monitor ambient humidity. Reproduced from ref. 77 with permission from Elsevier Ltd, copyright 2020. (e) The device prepared by PVDF/PI nanofiber film used for humidity sensing. Reproduced from ref. 128 with permission from Elsevier Ltd, copyright 2020.



2006, Wang's group<sup>48</sup> made the first realization of the conversion of mechanical energy to electrical energy in the nanoscale, which was the fundamental principle to prepare TENGs and PENGs for energy harvesting and powering small devices.<sup>75,129</sup> High-performance nanogenerators could also be integrated with sensor devices to create self-powered sensor devices.<sup>130</sup>

A flexible PENG that can power small portable and wearable electronics, such as watches and cell phones, has been prepared. It is expected to alleviate the need for miniaturized energy sources.<sup>99</sup> The power generation layer of this nanogenerator was made of PVDF and silane-modified KNN (SM-KNN) by electrospinning, and it was found that the PVDF/SM-KNN nanofiber film had good piezoelectric properties because the modified KNN reduced the coalescence of nanofillers and further induced the interaction between PVDF chains and KNN NRs to improve the crystallinity of the composite fiber. A high output voltage of  $\sim 21$  V, an output current of  $\sim 22$  mA, a current density of  $\sim 5.5$  mA cm<sup>-2</sup>, and a power density of  $\sim 15.5$  mW cm<sup>-2</sup> were produced by the PVDF/3% SM-KNN nanofiber film, which could be used to light an LED (Fig. 9a). A very flexible electrospun PVDF/PANI/g-C<sub>3</sub>N<sub>4</sub> blended nanocomposite fiber (PPBF) nanogenerator for piezoelectric energy harvesting was also reported.<sup>131</sup> The resultant PPBF nanogenerator had maximum output values of  $\sim 30$  V, 3.7  $\mu$ A, 14.7  $\mu$ W cm<sup>-2</sup>, and  $\sim 20\%$  for output voltage, output current, power density per unit area, and conversion efficiency. The results also demonstrated its excellent endurance and stability after more than 50 000 stress and de-stress cycles (Fig. 9b).

Compared to PENGs, TENGs have a better piezoelectric output and power efficiency due to their special structure. The TENG made by using electrospun PVDF/MXene and nylon 6/6 nanofiber films as the efficient negative frictional electric layer could deliver a peak power of 4.6 mW (power density of 11.213 W m<sup>-2</sup>) at a matched impedance of 2 M $\Omega$ ,<sup>132</sup> and also demonstrated outstanding stability at 60 000 contact separation cycles. More than 120 commercial red LEDs, a sports watch, or a thermohygrograph could be powered by the energy produced by the TENG (Fig. 9c). Besides, the cobalt-based nanoporous carbon (Co-NPC) combined with PVDF was also used as a highly electronegative friction material for improving the performance of TENGs in mechanical energy harvesting. At the Co-NPC content of 0.5 wt%, the maximum peak voltage of the PVDF/Co-NPC nanofiber film was 710 V, which was 2.1 times higher than that of pure PVDF.<sup>133</sup> A high-performance PVDF/Co-NPC-TENG in contact separation mode was created using the developed PVDF/Co-NPC nanofiber film in combination with the nylon-11 nanofiber film. The highest voltage, current density, and charge density of 948 V, 248.66 mA m<sup>-2</sup>, and 464  $\mu$ C m<sup>-2</sup> were obtained at a pressure of 50 N (Fig. 9d). Ambient humidity had a significant impact on the output performance of the TENG, and it was found that increasing the relative humidity would reduce the output voltage of the TENG. However, the PVDF/Co-NPC-TENG exhibited good charging capabilities for storage capacitors and could operate low-power electronics, such as digital stopwatches, at a high relative humidity level of 95%. To improve the triboelectric properties of the fiber film, liquid metal (LM) Galinstan nanodroplets were

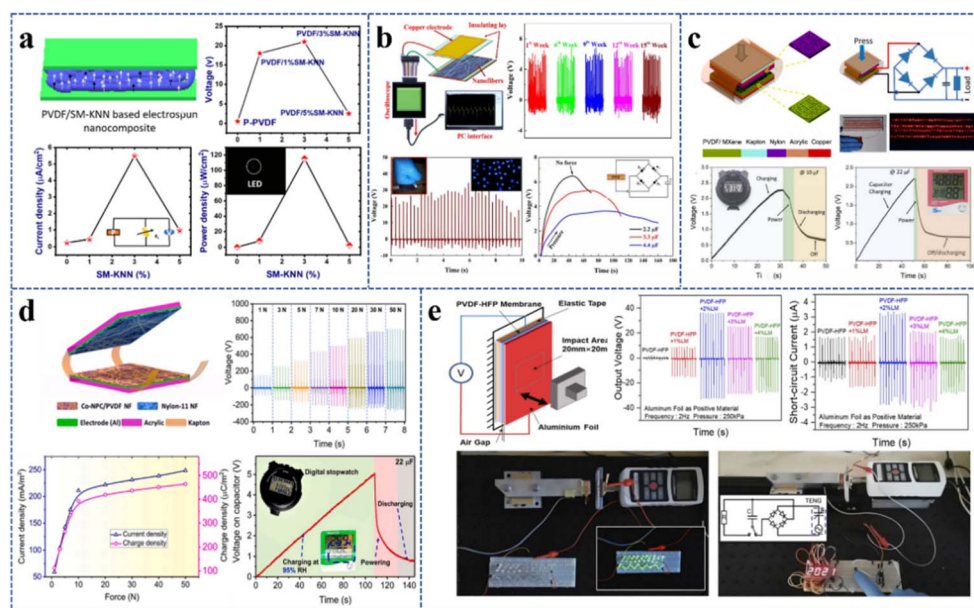


Fig. 9 Application of electrospun PVDF-based nanofiber films in the energy harvesting field: (a) PENG prepared from the PVDF/SM-KNN nanocomposite piezoelectric material. Reproduced from ref. 99 with permission from Elsevier Ltd, copyright 2020. (b) PVDF nanofibers with embedded PANI-graphitic carbon nitride nanosheet composites for piezoelectric energy conversion. Reproduced from ref. 131 with permission from the American Chemical Society, copyright 2019. (c) High-performance TENG based on MXene functionalized PVDF composite nanofibers. Reproduced from ref. 132 with permission from Elsevier Ltd, copyright 2020. (d) PVDF/Co-NPC nanofiber films for a high-performance TENG. Reproduced from ref. 133 with permission from Elsevier Ltd, copyright 2022. (e) Electrospun PVDF-HFP/LM nanofiber film with exceptional triboelectric performance. Reproduced from ref. 134 with permission from Elsevier Ltd, copyright 2021.



also added to prepare electrospun PVDF-HFP/LM nanofibers.<sup>134</sup> The TENG made of them was anticipated to provide power for smart devices as a sustainable energy source. PVDF-HFP added with 2 wt% LM was tested to have the highest peak voltage, current, and power density. At a shock frequency of 5 Hz and a shock pressure of 250 kPa, the measured voltage across the external resistor tended to stabilize at around 1680 V with a resistance of more than 500 M $\Omega$ . The electrical energy generated by the TENG could simultaneously light 80 LEDs and could also be successfully converted to DC and stored in a 22  $\mu$ F capacitor using a bridge rectifier circuit to illuminate a seven-segment LED display showing “2021” (Fig. 9e).

## 5. Summary and prospects

Electrospinning is a straightforward and economical method for the fabrication of polymeric nanofibers. Many researchers have been interested in electrospinning since it has given them a new option to create high-efficiency PVDF piezoelectric films. The strategies to improve the piezoelectric output performance of electrospun PVDF-based fiber films are primarily focusing on filler doping and structural design. The relationship between the doping filler content and the output voltage of PVDF-based composite fiber films has been reported in numerous publications. Lead-free piezoelectric ceramic materials, conductive materials, and some other inorganic and organic materials have relatively good effects on the output voltage of PVDF-based fiber films and are frequently used as doped fillers. Effective structural design, including a highly oriented structure, core-shell structure, and multilayer composite structure, can also improve the piezoelectric performance of PVDF-based fiber films to a certain extent. The mechanical properties have a significant impact on the performance and stability of the devices. It is possible to achieve higher mechanical strength by combining PVDF with polymers with better mechanical properties to prepare composite fibers, adding nanomaterials in the PVDF matrix for reinforcement, appropriately increasing the thickness of fiber films, and preparing fiber films with composite structures. The performance of the ultimate PVDF-based piezoelectric devices is expected to be further enhanced by the exploration of more efficient additives, structures, and combinations between them in the future. In-depth investigations of the synergistic design of materials and structures as well as the mix of various materials are essential. Although the preparation of electrospun PVDF-based piezoelectric fiber films and their applications are booming, there is currently no unified test standard for researchers to use when evaluating piezoelectric output, so that the comparability between various research endeavors is insufficient.

Electrospun PVDF-based fiber films can be used to prepare sensors, nanogenerators, and other piezoelectric devices, and they have been applied in a variety of fields because of their flexibility, lightweight, and biocompatible characteristics. Piezoelectric sensors have been applied in implantable sensing, human rehabilitation, motion monitoring, speech recognition, wind speed monitoring, and other aspects. Nanogenerators have been used for energy harvesting to provide sustainable

energy for smart homes or smart wearable devices. The sensors and nanogenerators based on electrospun PVDF fibers exhibit excellent performance and have great commercial value in the future. Although improved research outcomes have been achieved, the load-matching standard and minimum stability requirements of devices in practical applications need to be clarified, and more complete equipment integration is needed.

## Author contributions

Mengdi Zhang, Chengkun Liu, Boyu Li, Yutong Shen, and Hao Wang discussed the contents as well as wrote and reviewed the manuscript. Fenglei Zhou, Keyu Ji, Liang Wei, Xue Mao, and Runjun Sun edited, reviewed, and revised the manuscript. All authors have read and agreed to the published version of the manuscript.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

The authors gratefully acknowledge the financial support from the National Natural Science Foundation of China (51503168) and the Innovation Capability Support Plan of Shaanxi, China (2020PT-043).

## Notes and references

- Z. A. Alhassan, Y. S. Burezq, R. Nair and N. Shehata, *J. Nanomater.*, 2018, **2018**, 12.
- C. K. Jeong, J. H. Han, H. Palneedi, H. Park, G. T. Hwang, B. Joung, S. G. Kim, H. J. Shin, I. S. Kang, J. Ryu and K. J. Lee, *APL Mater.*, 2017, **5**, 074102.
- X. Chen, M. Han, H. Chen, X. Cheng, Y. Song, Z. Su, Y. Jiang and H. Zhang, *Nanoscale*, 2017, **9**, 1263.
- J. Huang, Y. Hao, M. Zhao, W. Li, F. Huang and Q. Wei, *ACS Appl. Mater. Interfaces*, 2021, **13**, 24774.
- S. M. S. Rana, M. T. Rahman, M. Salauddin, S. Sharma, P. Maharjan, T. Bhatta, H. Cho, C. Park and J. Y. Park, *ACS Appl. Mater. Interfaces*, 2021, **13**, 4955.
- T. Li, M. Qu, C. Carlos, L. Gu, F. Jin, T. Yuan, X. Wu, J. Xiao, T. Wang, W. Dong, X. Wang and Z. Q. Feng, *Adv. Mater.*, 2021, **33**, 2006093.
- Q. Zhou, T. Chen, S. Cao, X. Xia, Y. Bi and X. Xiao, *Appl. Phys. A*, 2021, **127**, 667.
- N. Shehata, A. H. Hassanin, E. Elnabawy, R. Nair, S. A. Bhat and I. Kandas, *Sensors*, 2020, **20**, 3111.
- C. Chang, V. H. Tran, J. Wang, Y. K. Fuh and L. Lin, *Nano Lett.*, 2010, **10**, 726.
- A. J. Lovinger, *Science*, 1983, **220**, 1115.
- X. Ma, J. Liu, C. Ni, D. C. Martin, D. B. Chase and J. F. Rabolt, *ACS Macro Lett.*, 2012, **1**, 428.
- P. Martins, A. C. Lopes and S. Lanceros-Mendez, *Prog. Polym. Sci.*, 2014, **39**, 683.



- 13 J. Fang, H. Niu, H. Wang, X. Wang and T. Lin, *Energy Environ. Sci.*, 2013, **6**, 2196.
- 14 L. Xie, G. Wang, C. Jiang, F. Yu and X. Zhao, *Crystals*, 2021, **11**, 644.
- 15 Y. Huang, G. Rui, Q. Li, E. Allahyarov, R. Li, M. Fukuto, G. J. Zhang, J. Z. Xu, Z. M. Li, P. L. Taylor and L. Zhu, *Nat. Commun.*, 2021, **12**, 675.
- 16 V. Sencadas, R. Gregorio and S. Lanceros-Méndez, *J. Macromol. Sci. B*, 2009, **48**, 514.
- 17 H. Pan, B. Na, R. Lv, C. Li, J. Zhu and Z. Yu, *J. Polym. Sci., Part B: Polym. Phys.*, 2012, **50**, 1433.
- 18 Z. H. Liu, C. T. Pan, L. W. Lin, J. C. Huang and Z. Y. Ou, *Smart Mater. Struct.*, 2014, **23**, 025003.
- 19 A. Lund, C. Gustafsson, H. Bertilsson and R. W. Rychwalski, *Compos. Sci. Technol.*, 2011, **71**, 222.
- 20 C. M. Wu, M. H. Chou and W. Y. Zeng, *Nanomaterials*, 2018, **8**, 420.
- 21 L. Persano, C. Dagdeviren, Y. Su, Y. Zhang, S. Girardo, D. Pisignano, Y. Huang and J. A. Rogers, *Nat. Commun.*, 2013, **4**, 1633.
- 22 W. Wang, J. Yang and Z. Liu, *IEEE Trans. Dielectr. Electr. Insul.*, 2017, **24**, 697.
- 23 Z. Yin, B. Tian, Q. Zhu and C. Duan, *Polymers*, 2019, **11**, 2033.
- 24 W. Yan, G. Noel, G. Loke, E. Meiklejohn, T. Khudiyev, J. Marion, G. Rui, J. Lin, J. Cherston, A. Sahasrabudhe, J. Wilbert, I. Wicaksono, R. W. Hoyt, A. Missakian, L. Zhu, C. Ma, J. Joannopoulos and Y. Fink, *Nature*, 2022, **603**, 616.
- 25 M. Du, L. Huang, J. Zheng, Y. Xi, Y. Dai, W. Zhang, W. Yan, G. Tao, J. Qiu, K. So, C. Ren and S. Zhou, *Adv. Sci.*, 2020, **7**, 2001410.
- 26 S. Qian, M. Liu, Y. Dou, Y. Fink and W. Yan, *Natl. Sci. Rev.*, 2022, **10**, 202.
- 27 S. Egusa, Z. Wang, N. Chocat, Z. M. Ruff, A. M. Stolyarov, D. Shemuly, F. Sorin, P. T. Rakich, J. D. Joannopoulos and Y. Fink, *Nat. Mater.*, 2010, **9**, 643.
- 28 S. Agarwal, A. Greiner and J. H. Wendorff, *Adv. Funct. Mater.*, 2009, **19**, 2863.
- 29 L. Zhang, J. Gui, Z. Wu, R. Li, Y. Wang, Z. Gong, X. Zhao, C. Sun and S. Guo, *Nano Energy*, 2019, **65**, 103924.
- 30 Z. Zhao, J. Li, X. Yuan, X. Li, Y. Zhang and J. Sheng, *J. Appl. Polym. Sci.*, 2005, **97**, 466.
- 31 T. Lei, L. Yu, L. Wang, F. Yang and D. Sun, *J. Macromol. Sci. B*, 2014, **54**, 91.
- 32 B. Zaarour, L. Zhu and X. Jin, *Soft Mater.*, 2019, **17**, 181.
- 33 S. Gee, B. Johnson and A. L. Smith, *J. Membr. Sci.*, 2018, **563**, 804.
- 34 H. Jiyong, G. Yuanyuan, Z. Hele, Z. Yinda and Y. Xudong, *J. Text. Inst.*, 2017, **109**, 843.
- 35 F. Huang, Q. Wei, J. Wang, Y. Cai and Y. Huang, *e-Polymers*, 2008, **8**, 1758.
- 36 N. Jia, Q. Xing, G. Xia, J. Sun, R. Song and W. Hang, *Mater. Lett.*, 2015, **139**, 212.
- 37 Y. Ahn, J. Y. Lim, S. M. Hong, J. Lee, J. Ha, H. J. Choi and Y. Seo, *J. Phys. Chem. C*, 2013, **117**, 11791.
- 38 N. Chamankar, R. Khajavi, A. A. Yousefi, A. Rashidi and F. Golestanifard, *Ceram. Int.*, 2020, **46**, 19669.
- 39 J. E. Lee, Y. E. Shin, G. H. Lee, J. Kim, H. Ko and H. G. Chae, *Composites, Part B*, 2021, **223**, 109098.
- 40 S. Sharafkhani and M. Kokabi, *AIP Conf. Proc.*, 2018, **1920**, 020001.
- 41 R. T. Selvan, C. Y. Jia, W. A. D. M. Jayathilaka, A. Chinappan, H. Alam and S. Ramakrishna, *Nano*, 2020, **15**, 2050049.
- 42 S. Lv, Y. Han, L. Shuai, B. Chen and J. Wan, *J. Lumin.*, 2021, **239**, 118303.
- 43 H. Lee, H. Kim, D. Y. Kim and Y. Seo, *ACS Omega*, 2019, **4**, 2610.
- 44 K. I. Park, M. Lee, Y. Liu, S. Moon, G. T. Hwang, G. Zhu, J. E. Kim, S. O. Kim, D. K. Kim, Z. L. Wang and K. J. Lee, *Adv. Mater.*, 2012, **24**, 2999.
- 45 Z. H. Lin, Y. Yang, J. M. Wu, Y. Liu, F. Zhang and Z. L. Wang, *J. Phys. Chem. Lett.*, 2012, **3**, 3599.
- 46 J. H. Jung, M. Lee, J. I. Hong, Y. Ding, C. Y. Chen, L. J. Chou and Z. L. Wang, *ACS Nano*, 2011, **5**, 10041.
- 47 J. H. Jung, C. Y. Chen, B. K. Yun, N. Lee, Y. Zhou, W. Jo, L. J. Chou and Z. L. Wang, *Nanotechnology*, 2012, **23**, 375401.
- 48 Z. L. Wang and J. H. Song, *Science*, 2006, **312**, 242.
- 49 F. R. Fan, W. Tang and Z. L. Wang, *Adv. Mater.*, 2016, **28**, 4283.
- 50 L. Csoka, I. C. Hoeger, O. J. Rojas, I. Peszlen, J. J. Pawlak and P. N. Peralta, *ACS Macro Lett.*, 2012, **1**, 867.
- 51 W. Wang, Y. Zheng, X. Jin, Y. Sun, B. Lu, H. Wang, J. Fang, H. Shao and T. Lin, *Nano Energy*, 2019, **56**, 588.
- 52 L. Yuan, W. Fan, X. Yang, S. Ge, C. Xia, S. Y. Foong, R. K. Liew, S. Wang, Q. V. Le and S. S. Lam, *Compos. Commun.*, 2021, **25**, 100680.
- 53 S. Anwar, M. H. Amiri, S. Jiang, M. M. Abolhasani, P. R. F. Rocha and K. Asadi, *Adv. Funct. Mater.*, 2020, **31**, 2004326.
- 54 V. Sencadas, C. Garvey, S. Mudie, J. J. K. Kirkensgaard, G. Gouadec and S. Hauser, *Nano Energy*, 2019, **66**, 104106.
- 55 J. Jiang, S. Tu, R. Fu, J. Li, F. Hu, B. Yan, Y. Gu and S. Chen, *ACS Appl. Mater. Interfaces*, 2020, **12**, 33989.
- 56 D. Wang, P. Shi, X. Li, P. Zhou, K. Zhao, Y. Wei, C. Jiang, J. Liang and R. A. Dorey, *Ceram. Int.*, 2018, **44**, 14258.
- 57 M. Koç, L. Paralı and O. Şan, *Polym. Test.*, 2020, **90**, 106695.
- 58 Y. Qi and M. C. McAlpine, *Energy Environ. Sci.*, 2010, **3**, 1275.
- 59 I. Choi, S. J. Lee, J. C. Kim, Y. Kim, D. Y. Hyeon, K. S. Hong, J. Suh, D. Shina, H. Y. Jeong and K. I. Park, *Appl. Surf. Sci.*, 2020, **511**, 145614.
- 60 M. Kumar, N. D. Kulkarni and P. Kumari, *Mater. Today: Proc.*, 2022, **56**, 1151.
- 61 S. Bairagi and S. W. Ali, *Energy Technol.*, 2019, **7**, 1900538.
- 62 N. Sinha, S. Goel, A. J. Joseph, H. Yadav, K. Batra, M. K. Gupta and B. Kumar, *Ceram. Int.*, 2018, **44**, 8582.
- 63 T. G. Smijs and S. Pavel, *Nanotechnol., Sci. Appl.*, 2011, **4**, 95.
- 64 N. A. Monteiro-Riviere, K. Wiench, R. Landsiedel, S. Schulte, A. O. Inman and J. E. Riviere, *Toxicol. Sci.*, 2011, **123**, 264.
- 65 K. Schilling, B. Bradford, D. Castelli, E. Dufour, J. F. Nash, W. Pape, S. Schulte, I. Tooley, J. Boschi and F. Schellauf, *Photochem. Photobiol. Sci.*, 2010, **9**, 495.



- 66 G. Y. Li, H. D. Zhang, K. Guo, X. S. Ma and Y. Z. Long, *Mater. Res. Express*, 2020, **7**, 095502.
- 67 X. Pu, J. W. Zha, C. L. Zhao, S. B. Gong, J. F. Gao and R. K. Y. Li, *Chem. Eng. J.*, 2020, **398**, 125526.
- 68 J. Li, S. Chen, W. Liu, R. Fu, S. Tu, Y. Zhao, L. Dong, B. Yan and Y. Gu, *J. Phys. Chem. C*, 2019, **123**, 11378.
- 69 Z. Liu, G. Li, Q. Qin, L. Mi, G. Li, G. Zheng, C. Liu, Q. Li and X. Liu, *Adv. Compos. Hybrid Mater.*, 2021, **4**, 1215.
- 70 R. Fu, S. Chen, Y. Lin, S. Zhang, J. Jiang, Q. Li and Y. Gu, *Mater. Lett.*, 2017, **187**, 86.
- 71 L. Huang, C. Lu, F. Wang and L. Wang, *RSC Adv.*, 2014, **4**, 45220.
- 72 A. Samadi, S. M. Hosseini and M. Mohseni, *Org. Electron.*, 2018, **59**, 149.
- 73 H. Yu, T. Huang, M. Lu, M. Mao, Q. Zhang and H. Wang, *Nanotechnology*, 2013, **24**, 405401.
- 74 O. M. Zeyrek, S. Oguzlar, E. C. Doluel, U. Kartal and M. Yurddaskal, *J. Mater. Sci.: Mater. Electron.*, 2019, **31**, 1960.
- 75 J. Yang, Y. Zhang, Y. Li, Z. Wang, W. Wang, Q. An and W. Tong, *Mater. Today Commun.*, 2021, **26**, 101629.
- 76 Q. Zhao, L. Yang, Y. Ma, H. Huang, H. He, H. Ji, Z. Wang and J. Qiu, *J. Alloys Compd.*, 2021, **886**, 161069.
- 77 S. Wang, H. Q. Shao, Y. Liu, C. Y. Tang, X. Zhao, K. Ke, R. Y. Bao, M. B. Yang and W. Yang, *Compos. Sci. Technol.*, 2021, **202**, 108600.
- 78 F. Mokhtari, M. Shamshirsaz, M. Latifi and J. Ferooghi, *Polymers*, 2020, **12**, 2697.
- 79 F. Mokhtari, M. Latifi, M. Shamshirsaz, M. Khelghatdoost and S. Rahmani, *J. Text. Inst.*, 2017, **108**, 1917.
- 80 C. K. Yen, K. Dutt, Y. S. Yao, W. J. Wu, Y. L. Shiue, C. T. Pan, C. W. Chen and W. F. Chen, *Polymers*, 2022, **14**, 331.
- 81 C. Li, H. Wang, X. Yan, H. Chen, Y. Fu and Q. Meng, *Coatings*, 2021, **11**, 1495.
- 82 C. Chen, Z. Bai, Y. Cao, M. Dong, K. Jiang, Y. Zhou, Y. Tao, S. Gu, J. Xu, X. Yin and W. Xu, *Compos. Sci. Technol.*, 2020, **192**, 108100.
- 83 F. Mokhtari, M. Shamshirsaz and M. Latifi, *Polym. Eng. Sci.*, 2016, **56**, 61.
- 84 J. Yang, J. Wang, Q. Zhang, F. Chen, H. Deng, K. Wang and Q. Fu, *Polymer*, 2011, **52**, 4970.
- 85 R. Neppalli, S. Wanjale, M. Birajdar and V. Causin, *Eur. Polym. J.*, 2013, **49**, 90.
- 86 Y. Xin, X. Qi, H. Tian, C. Guo, X. Li, J. Lin and C. Wang, *Mater. Lett.*, 2016, **164**, 136.
- 87 S. Shetty, A. Mahendran and S. Anandhan, *Soft Matter*, 2020, **16**, 5679.
- 88 S. Zhang and Q. An, *Chem. J. Chin. Univ.*, 2021, **42**, 1114.
- 89 B. Adeli, A. A. Gharehaghaji and A. A. A. Jeddi, *Iran. Polym. J.*, 2021, **30**, 535.
- 90 C. Merlini, A. Pegoretti, T. M. Araujo, S. D. A. S. Ramoaa, W. H. Schreiner and G. M. O. Barra, *Synth. Met.*, 2016, **213**, 34.
- 91 X. Cai, Z. Zheng, J. Xu, X. Tang and T. Lei, *J. Macromol. Sci. B*, 2016, **56**, 75.
- 92 A. Sengupta, S. Das, S. Dasgupta, P. Sengupta and P. Datta, *ACS Biomater. Sci. Eng.*, 2021, **7**, 1673.
- 93 R. S. Sabry and A. D. Hussein, *Polym. Test.*, 2019, **79**, 106001.
- 94 S. Bairagi and S. W. Ali, *Int. J. Energy Res.*, 2020, **44**, 5545.
- 95 D. Ponnammam, H. Parangusan, A. Tanvir and M. A. A. AlMa'adeed, *Mater. Des.*, 2019, **184**, 108176.
- 96 S. Bairagi and S. W. Ali, *Soft Matter*, 2020, **16**, 4876.
- 97 M. Hasanzadeh, M. R. Ghahhari and S. M. Bidoki, *J. Mater. Sci.: Mater. Electron.*, 2021, **32**, 15789.
- 98 K. Shi, B. Sun, X. Huang and P. Jiang, *Nano Energy*, 2018, **52**, 153.
- 99 S. Bairagi and S. W. Ali, *Energy*, 2020, **198**, 117385.
- 100 S. Banerjee, S. Bairagi and S. W. Ali, *Energy*, 2022, **244**, 123102.
- 101 Y. Su, C. Chen, H. Pan, Y. Yang, G. Chen, X. Zhao, W. Li, Q. Gong, G. Xie, Y. Zhou, S. Zhang, H. Tai, Y. Jiang and J. Chen, *Adv. Funct. Mater.*, 2021, **31**, 2010962.
- 102 D. Zhang, X. Zhang, X. Li, H. Wang, X. Sang, G. Zhu and Y. Yeung, *Eur. Polym. J.*, 2022, **166**, 110956.
- 103 T. Sharma, S. Naik, J. Langevine, B. Gill and J. X. J. Zhang, *IEEE Trans. Biomed. Eng.*, 2015, **62**, 188.
- 104 Y. M. Yousry, K. Yao, A. M. Mohamed, W. H. Liew, S. Chen and S. Ramakrishna, *Adv. Funct. Mater.*, 2020, **30**, 1910592.
- 105 Y. Jiang, L. Gong, X. Hu, Y. Zhao, H. Chen, L. Feng and D. Zhang, *Polymers*, 2018, **10**, 364.
- 106 N. Shehata, E. Elnabawy, M. Abdelkader, A. H. Hassanin, M. Salah, R. Nair and S. A. Bhat, *Polymers*, 2018, **10**, 965.
- 107 M. Zhu, M. Lou, I. Abdalla, J. Yu, Z. Li and B. Ding, *Nano Energy*, 2020, **69**, 104429.
- 108 D. Ponnammam D, M. M. Chamakh, A. M. Alahzm, N. Salim, N. Hameed and M. A. A. AlMaadeed, *Polymers*, 2020, **12**, 2344.
- 109 S. Wang, K. Shi, B. Chai, S. Qiao, Z. Huang, P. Jiang and X. Huang, *Nano Mater. Sci.*, 2021, **4**, 126.
- 110 T. Li, Z. Q. Feng, M. Qu, K. Yan, T. Yuan, B. Gao, T. Wang, W. Dong and J. Zheng, *ACS Nano*, 2019, **13**, 10062.
- 111 K. Maity and D. Mandal, *ACS Appl. Mater. Interfaces*, 2018, **10**, 18257.
- 112 Y. Li, M. Xu, Y. Xia, J. Wu, X. Sun, S. Wang, G. Hu and C. Xiong, *Chem. Eng. J.*, 2020, **388**, 124205.
- 113 S. Kalani, R. Kohandani and R. Bagherzadeh, *RSC Adv.*, 2020, **10**, 35090.
- 114 M. Kato and K. I. Kakimoto, *Mater. Lett.*, 2015, **156**, 183.
- 115 X. Wang, F. Sun, G. Yin, Y. Wang, B. Liu and M. Dong, *Sensors*, 2018, **18**, 330.
- 116 D. Chen, C. Wang, W. Chen, Y. Chen and J. X. J. Zhang, *Biosens. Bioelectron.*, 2015, **74**, 1047.
- 117 T. Li, Z. Q. Feng, K. Yan, T. Yuan, W. Wei, X. Yuan, C. Wang, T. Wang, W. Dong and J. Zheng, *J. Mater. Chem. B*, 2018, **6**, 5343.
- 118 R. F. Valentini, T. G. Vargo, J. A. G. Gardella Jr and P. Aebischer, *Biomaterials*, 1992, **13**, 183.
- 119 A. Bouaziz, A. Richert and A. Caprani, *Biomaterials*, 1997, **18**, 107.
- 120 Z. Yang, Z. Wu, D. Jiang, R. Wei, X. Mai, D. Pan, S. Vupputuri, L. Weng, N. Naik and Z. Guo, *J. Mater. Chem. C*, 2021, **9**, 2752.
- 121 H. F. Guo, Z. S. Li, S. W. Dong, W. J. Chen, L. Deng, Y. F. Wang and D. J. Ying, *Colloids Surf., B*, 2012, **96**, 29.





- 122 G. Y. Li, J. Li, Z. J. Li, Y. P. Zhang, X. Zhang, Z. J. Wang, W. P. Han, B. Sun, Y. Z. Long and H. D. Zhang, *Adv. Compos. Hybrid Mater.*, 2021, **5**, 766.
- 123 C. T. Pan, C. C. Chang, Y. S. Yang, C. K. Yen, Y. H. Kao and Y. L. Shiue, *Sens. Actuators, A*, 2020, **301**, 111708.
- 124 J. Zhao, F. Li, Z. Wang, P. Dong, G. Xia and K. Wang, *J. Mater. Sci.: Mater. Electron.*, 2021, **32**, 14715.
- 125 S. Garain, S. Jana, T. K. Sinha and D. Mandal, *ACS Appl. Mater. Interfaces*, 2016, **8**, 4532.
- 126 A. Sultana, M. M. Alam, S. K. Ghosh, T. R. Middy and D. Mandal, *Energy*, 2019, **166**, 963.
- 127 Q. Li, J. Xing, D. Shang and Y. Wang, *Sensors*, 2019, **19**, 1657.
- 128 M. Tayyab, J. Wang, J. Wang, M. Maksutoglu, H. Yu, G. Sun, F. Yildiz, M. Eginligil and W. Huang, *Nano Energy*, 2020, **77**, 105178.
- 129 H. Y. Mi, X. Jing, Q. Zheng, L. Fang, H. X. Huang, L. S. Turng and S. Gong, *Nano Energy*, 2018, **48**, 327.
- 130 J. Liu, B. Yang, L. Lu, X. Wang, X. Li, X. Chen and J. Liu, *Sens. Actuators, A*, 2020, **303**, 111796.
- 131 M. Khalifa and S. Anandhan, *ACS Appl. Nano Mater.*, 2019, **2**, 7328.
- 132 T. Bhatta, P. Maharjan, H. Cho, C. Park, S. H. Yoon, S. Sharma, M. Salauddin, M. T. Rahman, S. M. S. Rana and J. Y. Park, *Nano Energy*, 2021, **81**, 105670.
- 133 M. T. Rahman, S. M. S. Rana, M. A. Zahed, S. Lee, E. S. Yoon and J. Y. Park, *Nano Energy*, 2022, **94**, 106921.
- 134 Z. Sha, C. Boyer, G. Li, Y. Yu, F. M. Allieux, K. K. Zadeh, C. H. Wang and J. Zhang, *Nano Energy*, 2022, **92**, 106713.

