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1. Introduction

The rapid development of society and the excessive use of fuel due to globalization and industrialization have not only aggravated the emission of carbon dioxide (CO_2), but have also posed a serious threat to the ecosystem of earth. CO_2 is considered to be the main greenhouse gas causing global warming. In the last few decades, CO_2 has accumulated in the atmosphere in large amounts.¹⁻⁵ Through photoreduction CO_2 technology, CO_2 can be effectively converted into fuels or more valuable chemicals, which means that it can not only reduce the greenhouse effect caused by CO_2 , but can also fulfill the needs of looking for new types of energy.⁶⁻¹¹ However, most photocatalysts currently developed are restricted by a narrow light absorption range and poor photo-generated carrier separation efficiency, resulting in low photocatalytic activity efficiency.

Compared with single-component catalysts, multicomponent composite photocatalysts are more conducive to improving charge separation and photocatalytic activity.

Europium single atom based heterojunction photocatalysts with enhanced visible-light catalytic activity[†]

Yini li, Yang Qu D and Guofeng Wang *

Promoting charge separation is still the main challenge in the rational design of photocatalysts for visible light photocatalytic reduction of CO₂. Based on the unique advantages of the Eu³⁺ single atom and the Type-II heterojunction in the field of photocatalytic reduction, a Eu^{3+} single atom doped CdS/InVO₄ Type-II heterojunction was successfully constructed through the combination of hydrothermal and in situ synthesis. The prepared CdS/InVO₄: Eu^{3+} single atom composite material has excellent photocatalytic CO₂ reduction activity under visible light. The XPS test results show that the In 3d, O 1s, V 2p, and Eu 3d peaks of 10-CdS/IVO:Eu³⁺ move slightly towards a low energy direction, while, the positions of Cd 3d and S 2p peaks are both shifted to the direction of high binding energies, which indicated that the electrons can be transferred from CdS to IVO with the help of Eu³⁺. The CO yield of CdS/InVO₄:Eu³⁺ was 11.02 μ mol q⁻¹ h⁻¹, which was almost 2.4 times that of pure InVO₄ and 1.9 times that of pure CdS, and the yield of CH₄ of CdS/InVO₄:Eu³⁺ reached 8.24 μ mol g⁻¹ h⁻¹, which was 5.3 times and 10.6 times that of $InVO_4$ and CdS, respectively. Both the theoretical and experimental results show that the single atom Eu^{3+} can not only modify InVO₄ and activate CO₂ molecules, but can also promote charge separation between the interface and improve the photocatalytic performance. It is expected that the research results of this paper can provide a novel idea for the rational design of rare earth monatomic catalysts.

Therefore, the reasonable construction of semiconductor heterojunction photocatalysts is expected to improve the photocatalytic efficiency of the catalyst.¹²⁻¹⁶ The heterojunction with good semiconductor energy level matching and close contact can accelerate charge separation and extend the lifetime of electrons.¹⁷⁻²⁴

In recent years, indium vanadate (InVO₄) has been considered as a photocatalyst with great potential because of its appropriate band gap energy (2.15 eV) and high chemical stability.^{25–28} However, the recombination rate of electron–hole pairs of InVO₄ is relatively high, which limits the improvement of photocatalytic performance. As another excellent catalyst material, CdS has been widely used in hydrogen production from water splitting and photocatalytic degradation of organic pollutants.^{29–33} CdS has high visible light response and carrier transport ability. More importantly, the suitable band gap (2.4 eV) of CdS can be well matched with that of InVO₄; so it is possible to form a Type-II heterostructure. How to greatly improve charge transfer at the heterogeneous interface is still the focus of current research work.^{34–36}

Atomically thin semiconductors can provide a large number of low-coordination surface atoms, which can be used as the catalytic center for CO_2 reduction to improve the photocatalytic activity. The ultra-thin thickness can also shorten the distance of charge transfer, so that the photo-generated carriers can

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more easily migrate from the inside to the surface.^{37–39} Particularly, single-atom catalysts have attracted extensive attention, in which each atom is used as an active center, so the efficiency of catalysts is greatly improved.^{40–44} However, there are few studies on rare earth (RE) single-atom catalysts. It is well known that RE ions have special energy level structures and have been widely used in the modification of catalysts, but the possible mechanism to improve the catalytic performance of RE ions is yet to be explored.^{45–51}

Herein, $InVO_4$ sheets with a uniform thickness of less than 10 nm were successfully synthesized, and were further compounded with CdS nanoparticles to form a Type-II heterojunction with enhanced visible-light catalytic activity. This is the first report of europium single atom photocatalytic materials (Table S1[†]). The prepared CdS/InVO₄:Eu³⁺ single atom composite material has excellent photocatalytic CO₂ reduction activity under visible light.

2. Experimental

The chemicals and characterization details are shown in the ESI. \dagger

2.1 Synthesis of CdS nanoparticle

4 mmol (Cd(CH₃COO)₂·2H₂O) was dissolved in 20 mL dodecylamine, and then 4 mmol S powder was added into the above solution under constant magnetic stirring. After further stirring for 30 minutes at 50 °C, the mixture was transferred to a Teflon lined stainless steel autoclave with an internal volume of 50 mL and heated at 200 °C for 24 h; the precipitate obtained by the reaction was repeatedly washed with ethanol and CS₂ solvent, centrifuged, and dried at room temperature to obtain small CdS nanoparticles.

2.2 Synthesis of InVO₄ (IVO) and InVO₄:Eu³⁺ (IVO:Eu³⁺)

1.0 mmol Na₃VO₄·12H₂O and 1.0 mmol InCl₃·4H₂O were dissolved in 20 and 10 mL deionized water, respectively. The InCl₃ aqueous solution was added into the Na₃VO₄ solution under constant magnetic stirring, and then, the pH value of the solution was adjusted to about 1–2. The solution was further transferred to a Teflon-lined stainless steel autoclave and kept at 180 °C for 18 h in the oven. After cooling to room temperature, the sediment at the bottom was filtered, washed with deionized water several times, freeze-dried in a freeze dryer, and then ultra-thin indium IVO (InVO₄) nanosheets were successfully obtained. The synthesis process of IVO:Eu³⁺ (InVO₄:Eu³⁺) is similar to that of IVO, except that 1 mol% InCl₃ is replaced by EuCl₃ in the initial reaction process.

2.3 Synthesis of CdS/IVO:Eu³⁺ and CCdS/IVO:Eu³⁺

First, CdS small particles were synthesized by a hydrothermal method, and then CdS/IVO:Eu³⁺ was obtained by adding CdS particles into the precursor solution of synthetic IVO:Eu³⁺. According to the ratio of CdS to IVO:Eu³⁺, these samples were named 5-CdS/IVO:Eu³⁺, 7-CdS/IVO:Eu³⁺, 10-CdS/IVO:Eu³⁺, and 13-CdS/IVO:Eu³⁺. Similarly, CCdS/IVO:Eu³⁺ was obtained by

adding CCdS (commercial CdS) particles into the precursor solution of synthetic IVO:Eu³⁺. The synthesis parameters and sample abbreviations are shown in Table S2.[†]

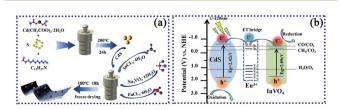
3. Results and discussion

3.1 Overall design route of the work

As mentioned above, the Eu³⁺ single atom doped CdS/InVO₄ Type-II heterojunction can be constructed through the combination of a hydrothermal method and *in situ* synthesis method. Here, the CCdS/InVO4 Type-II heterojunction was used as a reference sample to study the photocatalytic performance. Scheme 1a shows the synthesis diagrams of CdS and CdS/ IVO:Eu³⁺. The photoreduction process of CO₂ under visible light and the proposed photocatalytic mechanism are shown in Scheme 1b. In this study, the ⁵D₁ level of Eu³⁺ is located between the conduction bands of CdS and IVO, which is suitable for forming a charge transfer bridge and promoting charge separation. Thus, the single-atom Eu³⁺ can not only modify IVO by doping, but can also serve as an interface charge transfer bridge between IVO and CdS, which fully improves the charge separation between the interfaces and improves the photocatalytic activity of CO₂ reduction.

3.2 Materials characterization

As expected, a series of Eu³⁺ single atom based CdS/IVO:Eu³⁺ and CCdS/IVO:Eu³⁺ composite materials were successfully synthesized. As mentioned above, the synthesis parameters and sample abbreviations are shown in Table S2.[†] The scanning electron microscope (SEM) and transmission electron microscopy (TEM) images of CdS, IVO, IVO:Eu³⁺, and 10-CdS/IVO:Eu³⁺ are shown in Fig. 1a-f. The SEM and TEM images of IVO indicate that IVO is mainly composed of smooth, flexible, and sheetlike nanostructures, and a small amount of Eu³⁺ doping (1 mol%) has little effect on the morphology. For the 10-CdS/ IVO:Eu³⁺ sample, it can be observed that the nanosheet structure of IVO:Eu³⁺ is still retained (Fig. 1c, g-i), and the HRTEM image shows that the lattice fringe at a distance of 0.27 nm between planes belongs to the (112) crystal plane of the orthogonal phase of the IVO. The dispersion state of the Eu element in CdS/IVO:Eu³⁺ was further verified by aberrationcorrected HAADF-STEM (AC HAADF-STEM) measurements (Fig. 1g-p). Clear bright spots can be observed in the AC HAADF-STEM image of 10-CdS/IVO:Eu³⁺, indicating that the Eu element exists in the composite material as a single atom. The energy



Scheme 1 (a) Synthesis diagrams of CdS and CdS/IVO: Eu^{3+} . (b) The proposed mechanism diagram of photocatalysis and the photoreduction of CO₂ process under visible light irradiation.

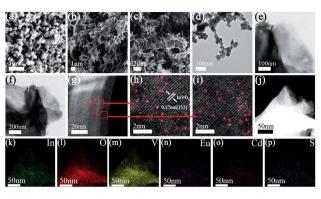


Fig. 1 SEM and TEM images of (a and d) CdS, (b and e) IVO, (c) 10-CdS/ IVO:Eu³⁺, and (f) IVO:Eu³⁺. (g-i) AC HAADF-STEM images of 10-CdS/ IVO:Eu³⁺. (j-p) HAADF-STEM images and EDX elemental mappings of 10-CdS/IVO:Eu³⁺.

dispersive spectroscopy (EDX) spectra showed the same results as expected, indicating that In, O, V, Eu, Cd, and S elements are uniformly distributed in the composite material. In addition, the SEM images of CCdS and 10-CCdS/IVO:Eu³⁺ show that CCdS is mainly composed of irregular and amorphous nanocrystalline structures, and a large amount of aggregation occurs in 10-CCdS/IVO:Eu³⁺ composites, as shown in Fig. S1.† The atomic force microscope (AFM) images and the corresponding height cross section profiles of IVO and 10-CdS/IVO:Eu³⁺ are illustrated in Fig. S2.† The thickness of the nanosheet was measured from the cross section height profiles, which reveals that the average thicknesses of IVO and 10-CdS/IVO:Eu³⁺ are 6.8 and 7.0 nm, respectively.

The crystal structures of the prepared samples were studied by XRD characterization (Fig. 2a). The results of XRD patterns show that the pure IVO can be easily indexed to the orthorhombic phase (JCPDS no. 48-0898); the obvious and sharp peaks indicate that IVO is well crystallized. In the XRD diffraction peaks of IVO:Eu³⁺, no obvious Eu³⁺ diffraction peaks were observed, indicating that the atomic-level dopant Eu³⁺ ions are well combined with the IVO lattice. For CdS/IVO:Eu³⁺, the characteristic peaks of IVO:Eu³⁺ and CdS were detected, proving that there is a two-phase composition in the CdS/IVO:Eu³⁺

(a) (b)1603 LCAS/IVO-F 5-CdS/IVO:Eu VO:Eu³ IVO 1610cm 0 2500 2000 1500 20 (degree) Energy (keV) Wavenumber (cm⁻¹) (d) e IVO IVO:Eu³ =133.58 Sarr=179.96m2g =334 81m²e ahv)2 0.2 0.4 0.6

Fig. 2 (a) XRD patterns, (b) EDS, (c) FT-IR spectra, (d) UV-vis absorption spectra, (e) plot of $(ahv)^2$ versus photon energy, and (f) N₂ adsorption–desorption isotherm curves of CdS, IVO, IVO:Eu³⁺ and CdS/IVO:Eu³⁺.

heterojunction. As the content of CdS increases, the peak intensity of CdS also increases. The energy dispersive spectroscopy (EDS) spectra of 10-CdS/IVO:Eu³⁺ further prove the formation of a heterojunction, in which the signals of In, V, O, Eu, Cd, and S elements can be clearly observed (Fig. 2b). In order to better evaluate the effect of Eu content on photocatalytic performance, we carried out the ICP test of different ratios of CdS/IVO:Eu³⁺. Among them, the Eu mass concentrations of 7-CdS/IVO:Eu³⁺, 10-CdS/IVO:Eu³⁺, and 13-CdS/IVO:Eu³⁺ are 1.28%, 1.27%, and 1.21%, respectively. It is noted that during the synthesis of CdS/IVO:Eu³⁺, the concentration of Eu³⁺ in IVO:Eu³⁺ is fixed, and the concentration of Eu³⁺ in CdS/IVO:Eu³⁺ changes with the different proportions of CdS and IVO:Eu³⁺. On the whole, the Eu³⁺ concentration increases with the increase of IVO:Eu³⁺ content.

The Fourier transform infrared (FT-IR) spectra of pure IVO, IVO:Eu³⁺, 5-CdS/IVO:Eu³⁺, 10-CdS/IVO:Eu³⁺, and CdS are shown in Fig. 2c. For the CdS/IVO:Eu³⁺ composite material, almost the same characteristic peaks as IVO can be observed, of which 764, 925, and 1014 cm⁻¹ are assigned to V–O–In bridging and V–O stretching, respectively. A weaker Cd–S vibration (2861 and 2920 cm⁻¹) was observed in the 3000–2800 cm⁻¹ area. In addition, the position of the VO^{4–} group (1603 cm⁻¹) for CdS/IVO:Eu³⁺ has a weaker blue shift relative to that of IVO or IVO:Eu³⁺ (1610 cm⁻¹), which can be attributed to the interaction of IVO:Eu³⁺ and CdS.

The UV-vis absorption spectra of the obtained samples were measured by UV-vis absorption spectroscopy; all samples respond under ultraviolet and visible light (Fig. 2d). Compared with pure IVO, the absorption range of the Eu³⁺ doped one broadens toward the long wave direction. Meanwhile, the visible light absorption of CdS/IVO:Eu³⁺ was significantly enhanced, and the absorption edge showed a red shift with the increase of CdS content. The bandgap energies (E_g) of IVO, IVO:Eu³⁺, 10-CdS/IVO:Eu³⁺, and CdS are estimated to be about 2.09, 2.01, 1.98 and 2.42 eV, respectively (Fig. 2e and S3[†]). The reduction in the band gap of the composite material may be caused by the interaction of chemical bonds between semiconductors and the Eu³⁺ doping. The thermogravimetric analysis of 10-CdS/IVO:Eu³⁺ is shown in Fig. S4.† The weightlessness stage mainly occurs below 100 °C, which is caused by the evaporation of residual solvent. The composite materials show no obvious frame collapse, which proves that it has good thermal stability.

The hysteresis loops of the nitrogen adsorption and desorption isotherm belong to the second type isotherm (Fig. 2f), indicating that the samples have large pores. The specific surface areas of the composite samples 5-CdS/IVO:Eu³⁺ and 10-CdS/IVO:Eu³⁺ are as high as 334.81 and 355.23 m² g⁻¹, which are significantly larger than those of IVO and CdS. In general, the larger the surface area is, the more conducive it is to the adsorption of CO₂, thereby improving the photocatalytic performance.

The chemical compositions and the binding energy of chemical bonds of the as-prepared 10-CdS/IVO:Eu³⁺ photocatalyst were further analyzed by X-ray photoelectron spectroscopy (XPS). The results of XPS spectra show that 10-CdS/

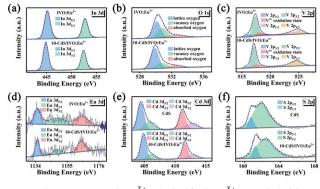


Fig. 3 XPS spectra of IVO: Eu^{3+} , 10-CdS/IVO: Eu^{3+} , and CdS. (a) In 3d, (b) O 1s, (c) V 2p, (d) Eu 3d, (e) Cd 3d, and (f) S 2p.

IVO:Eu³⁺ is mainly composed of In, O, V, Eu, Cd, and S elements (Fig. S5^{\dagger}). In the In 3d XPS spectrum (Fig. 3a) of IVO:Eu³⁺, the two peaks at 445.2 and 452.7 eV are attributed to the binding energies of In $3d_{5/2}$ and In $3d_{3/2}$, indicating the existence of In³⁺ species. After compounding with CdS, the peak positions of In 3d_{5/2} and In 3d_{3/2} shift to 444.9 and 452.4 eV, respectively. Obviously, the peak positions of In $3d_{5/2}$ and $3d_{3/2}$ move slightly towards the low energy direction, indicating that the charge density around In increases. Compared with IVO:Eu³⁺, the O 1s peaks for 10-CdS/IVO:Eu³⁺ shifted from 530.1 eV (lattice oxygen), 530.8 eV (vacancy oxygen), and 532.4 eV (absorbed oxygen) to 529.9, 530.7, and 532.3 eV, respectively (Fig. 3b). Among them, the formation of vacancy oxygen in IVO is necessary to maintain a charge balance when V⁵⁺ converts to V³⁺ and V⁵⁺ to V⁴⁺.⁵² Fig. 3c shows the V 2p spectra of 10-CdS/ IVO:Eu³⁺ and IVO:Eu³⁺. For IVO:Eu³⁺, the peaks at 516.2 and 523.6 eV correspond to V $2p_{3/2}$ and V $2p_{1/2}$ of V³⁺ ions, while the peaks at 517.4 and 525.0 eV are attributed to V $2p_{3/2}$ and V $2p_{1/2}$ of V⁵⁺ ions, and the other peak at about 517.9 eV can be indexed to the oxidation state of V⁴⁺, respectively. For 10-CdS/IVO:Eu³⁺, the location of V $2p_{3/2}$ and V $2p_{1/2}$ peaks of V³⁺ ions shift to 515.9 and 522.1 eV, the location of V $2p_{3/2}$ and V $2p_{1/2}$ peaks of V⁵⁺ ions shift to 517.1 and 524.6 eV, and the location of oxidation state of the V⁴⁺ peak shifts to 517.7 eV, respectively. Compared with IVO:Eu³⁺, the Eu 3d peaks of 10-CdS/IVO:Eu³⁺ shifted from 1135.4 eV (Eu 3d_{5/2}), 1149.5 eV (Eu 3d_{3/2}), and 1164.8 eV (Eu 3d_{3/} 2) to 1135.4, 1148.7, and 1164.4 eV, respectively (Fig. 3d). The Cd 3d spectrum of CdS displays four peaks at 404.5, 405.8, 411.3, and 412.5 eV, which exactly belonged to Cd 3d_{5/2} and Cd 3d_{3/2} of Cd^{2+} (Fig. 3e). After compounding with IVO:Eu³⁺, the Cd $3d_{5/2}$ peaks shifted to 404.9 and 406.2 eV, and the Cd 3d_{3/2} peaks shifted to 411.6 and 412.9 eV, respectively. In the spectra of S 2p (Fig. 3f), the S $2p_{3/2}$ (160.7 eV) and S $2p_{1/2}$ (161.8 eV) peaks also shifted to the high binding energy side (161.2 eV and 162.3 eV) after forming the 10-CdS/IVO:Eu³⁺ heterojunction. Interestingly, it is found that the In 3d, O 1s, V 2p, and Eu 3d peaks of 10-CdS/IVO:Eu³⁺ move slightly towards the low energy direction, indicating that the charge density around In, O, V, and Eu increases. However, the positions of Cd 3d and S 2p peaks are both shifted to the direction of high binding energy, which indicated that the charge density around Cd and S decreases.

These results further demonstrate that Eu^{3+} exists as a charge transfer bridge between CdS and IVO: Eu^{3+} , and the electrons can transfer from CdS to IVO, which promotes charge separation and improves photocatalytic activity. The charge separation will be also discussed in detail with subsequent theoretical calculations.

3.3 Photocatalytic properties

The photocatalytic performances of IVO, IVO:Eu³⁺, CdS, CCdS, CdS/IVO, CdS/IVO:Eu³⁺, CCdS/IVO, and CCdS/IVO:Eu³⁺ under visible light are shown in Fig. 4. As predicted, the photocatalytic performance of CCdS/IVO:Eu³⁺ was significantly improved compared with those of pure IVO, IVO:Eu³⁺ and CCdS. And the photocatalytic performance of CdS/IVO:Eu³⁺ was also significantly improved compared with those of pure IVO, IVO:Eu³⁺ and CdS. Particularly, the photocatalytic performance of CdS/ IVO:Eu³⁺ was better than that of CCdS/IVO:Eu³⁺. The CO and CH₄ production rates of 10-CCdS/IVO:Eu³⁺ reached 7.01 and 5.61 μ mol g⁻¹ h⁻¹, respectively. The CO content is equivalent to 1.56 times that of IVO (4.52 μ mol g⁻¹ h⁻¹) and 1.63 times that of CCdS (4.3 μ mol g⁻¹ h⁻¹). The CH₄ content for 10-CCdS/IVO:Eu³⁺ is equivalent to 3.6 times that of IVO (1.56 μ mol g⁻¹ h⁻¹) and 10.6 times that of CCdS (0.53 μ mol g⁻¹ h⁻¹). The CO production rates of 10-CdS/IVO:Eu³⁺ reached 11.02 µmol g⁻¹ h⁻¹, which is approximately 2.4 times that of IVO (4.52 μ mol g⁻¹ h⁻¹) and 1.9 times that of CdS (5.7 μ mol g⁻¹ h⁻¹). The CH₄ production rates of 10-CdS/IVO:Eu³⁺ reached 8.24 μ mol g⁻¹ h⁻¹, which is 5.3 times higher than that of pure IVO (1.56 μ mol g⁻¹ h⁻¹) and 10.6 times higher than that of CdS (0.78 μ mol g⁻¹ h⁻¹). In order to investigate the CO₂ adsorption capacity of the as-prepared samples, the CO2-TPD measurements of the samples were performed, as shown in Fig. S6.[†] The CO₂-TPD profile of 10-CdS/ IVO: Eu^{3+} shows that there is a strong CO_2 desorption band in the range of 200-350 °C, implying that the surface of 10-CdS/ IVO:Eu³⁺ could induce a certain extent of CO₂ adsorption. It is worth noting that the CO₂ desorption peak area of 10-CdS/ IVO:Eu³⁺ is much larger than that of IVO, so the interaction between CO₂ and 10-CdS/IVO:Eu³⁺ is much stronger.

The O₂, CO, CH₄, and H₂ photocatalytic production amounts of IVO, 10-CdS/IVO:Eu³⁺ and CdS are shown in Fig. S7.† According to our test results, the numbers of electrons consumed in the reduction reactions of IVO, 10-CdS/IVO:Eu³⁺

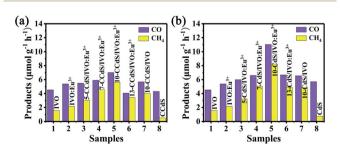


Fig. 4 Production of CO and CH₄ from the CO₂ photoreduction systems without sacrificial agents for the photocatalytic reaction. (a) IVO, IVO:Eu³⁺, CCdS/IVO:Eu³⁺, CCdS/IVO, and CCdS. (b) IVO, IVO:Eu³⁺, CdS/IVO:Eu³⁺, CdS/IVO, and CdS.

and CdS are 24.56, 93.16, and 20.64 μ mol g⁻¹ h⁻¹, and the numbers of holes consumed on the water oxidation reaction are 24.72, 94.00, and 21.20 μ mol g⁻¹ h⁻¹, respectively. The calculation results are basically in line with the balance of the number of electrons consumed in the reduction reaction and the number of holes consumed in the water oxidation reaction, and are basically consistent with $4 \times O_2 = 2 \times CO + 8 \times CH_4 + 2$ \times H₂. In addition, the photocatalytic activity of 10-CdS/IVO:Eu³⁺ is relatively stable (Fig. S8[†]). It is found that the crystal phase of the sample has no obvious change after the photocatalysis test, but the morphology of the sample shows that the sample has agglomeration which further proves that the photocatalyst has good stability (Fig. $S9^+$). To prove the role of Eu^{3+} , the photocatalytic performance between CdS/IVO:Eu3+ and CdS/IVO (or between CCdS/IVO:Eu³⁺ and CCdS/IVO) were compared. As expected, the photocatalytic performance of CdS/IVO:Eu³⁺ (or CCdS/IVO:Eu³⁺) was improved compared with that of pure CdS/ IVO (or CCdS/IVO). We suggested that the Eu³⁺ single atom plays multiple roles in the CdS/IVO:Eu³⁺ (or CCdS/IVO:Eu³⁺) samples, which will be discussed in the later theoretical calculation in detail.

Fig. 5a shows the transient photocurrent test results of IVO:Eu³⁺, CdS and CdS/IVO:Eu³⁺. Compared with IVO:Eu³⁺ and CdS, CdS/IVO:Eu³⁺ has significantly higher photocurrent intensity, and the photocurrent intensity of 10-CdS/IVO:Eu³⁺ is the highest. We suggested that CdS/IVO:Eu³⁺ has a higher photocatalytic activity by reducing the recombination rate of photoinduced charge carriers and promoting the separation and transfer of photogenerated electron–hole pairs, which is consistent with the results of photocatalytic performance. The electrochemical impedances (EIS) results are shown in Fig. 5b; it is well known that the smaller the arc radius, the faster the interface charge transfer, indicating that the charge transfer of 10-CdS/IVO:Eu³⁺ is the fastest. The coumarin fluorescence method was employed to explore the 'OH produced by the

(b)³⁰⁰⁰

(sundo) miZ 100

20

160

120

Intensity (a.u.)(p)

IVO:Eu³

Zre (ohms)

10-CdS/IVO:Eu

100 Time (us)

600

Wavelength (nm)

1600

700

7-CdS/IVO·Eu

400 800 1200

10-CdS/IVO:H

500

IVO:Eu3

- CdS - IVO:Eu

7-CdS/IVO:Eu3

10-CdS/IVO:Eu3

250 300

IVO:Eu³⁴

10-CdS/IVO:Eu

Time (s)

450

Wavelength (nm)

350

Fig. 5 (a) Transient photocurrent curves, (b) electrochemical impedance spectra (EIS) in the light, (c) spectral changes of hydroxyl radicals produced under 370 nm light excitation, and (d) photoluminescence spectra and luminescence decay curves obtained under 370 nm light excitation of CdS, IVO:Eu³⁺, and CdS/IVO:Eu³⁺.

480

samples during the photocatalytic reduction of CO₂, and the fluorescence spectra of 7-hydroxycoumarin were measured under 370 nm light excitation (Fig. 5c). The fluorescence spectrum signal of 10-CdS/IVO:Eu³⁺ is much higher than those of IVO:Eu³⁺ and CdS, indicating that 10-CdS/IVO:Eu³⁺ produces more 'OH and higher photoactivity during the reduction process. Fig. 5d presents the photoluminescence spectra of different samples. Here, the emission band positions of CdS and IVO:Eu3+ are obviously different. However for CdS/ IVO:Eu³⁺, only the emission band of IVO:Eu³⁺ was observed, indicating that the electrons can transfer from CdS to IVO. The illustration in Fig. 5d shows the luminescence decay curves of different samples. It is noted that the monitoring wavelengths are 440, 540, and 540 nm for CdS, IVO:Eu3+, and 10-CdS/ IVO:Eu³⁺, respectively. The average fitting data of the fluorescence lifetimes of CdS, IVO:Eu3+, and 10-CdS/IVO:Eu3+ are 173.40, 147.66 and 150.90 us, respectively. Compared with IVO:Eu³⁺, the fluorescence lifetime of 10-CdS/IVO:Eu³⁺ was improved, which further proves that the energy is transferred from CdS to IVO:Eu³⁺, and the formation of the heterojunction can enhance electron transport and reduce hole-electron pair recombination.

3.4 DFT calculations and discussion

To further prove that the CdS/IVO:Eu³⁺ is conducive to charge separation, DFT was used to optimize the crystal structures and calculate the deformation charge densities of CdS, IVO, and IVO:Eu³⁺. The calculation parameters for the (112) surface of samples with different section thicknesses and vacuum layer thicknesses and the corresponding sample names are shown in Table S3.[†] The geometric structures and deformation charge densities of CdS, IVO, and IVO:Eu³⁺ are shown in Fig. 6 and S10.[†] The lattice parameters of CdS are a = 6.26 Å, b = 6.26 Å, and c = 7.16 Å. The lattice parameters of IVO are a = 5.55 Å, b = 5.55 Å, and c = 7.18 Å. The lattice parameters of IVO are a = 5.55 Å, b = 5.35 Å, b = 5.35 Å, and c = 7.41 Å. The adsorption energies (E_{ads}) of CO₂ on the (112) surfaces of CdS (thin), IVO (thin), and IVO:Eu³⁺ (thin) are also shown in Fig. 6. The values of E_{ads} are 0.5, 0.3, and 0.7 eV for CdS (thin), IVO (thin), and IVO:Eu³⁺

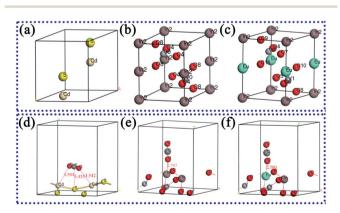


Fig. 6 The geometric structures of (a) CdS, (b) IVO, and (c) IVO:Eu³⁺. The adsorption energy (E_{ads}) of CO₂ on the (112) surface of (d) CdS (thin), (e) IVO (thin), and (f) IVO:Eu³⁺ (thin).

 $(a)_{1600}$

> 04 420

150

200

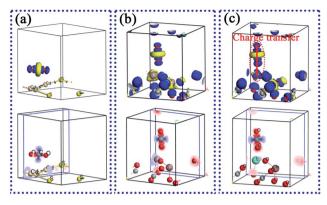


Fig. 7 Electron density difference (the blue area shows that the electron density is increasing; in contrast, the yellow area shows that the electron density is decreasing) of CO_2 adsorbed on the (112) surface of the (a) CdS (thin), (b) IVO (thin), and (c) IVO:Eu³⁺ (thin).

(thin), respectively. The results show that the bond length of O-Eu is shorter than that of O-In, and the adsorption energy of CO_2 on the (112) surface of IVO (thin) is larger than that of IVO:Eu³⁺ (thin). Noteworthily, CO_2 adsorption energy is also related to the location of CO_2 adsorption, as shown in Fig. S11.[†] The charge density differences of CO_2 adsorbed on the (112) surface of the CdS (thin), IVO (thin), and IVO:Eu³⁺ (thin) are shown in Fig. 7. The geometric structure and charge density difference of CdS/IVO (thin), CdS/IVO:Eu³⁺ (thin), and CdS/IVO:Eu³⁺ (thick) heterojunctions are shown in Fig. 8. All the results indicate that the existence of the Eu³⁺ single atom is conducive to the activation of CO_2 molecules and the interfacial charge transfer.

The calculated Fermi levels, vacuum levels, and work functions of the (112) surface of samples with different section thicknesses and vacuum layer thicknesses are shown in Fig. 9. The results indicate that the values of work function are 5.940, 4.417, 7.184, 3.521, 6.797, 5.444, 3.767, 4.187, and 4.874 eV for CdS (thin), CdS (thick), IVO (thin), IVO (thick), IVO–Eu (thin), IVO–Eu (thick), CdS/IVO:Eu³⁺ (thin), CdS/IVO:Eu³⁺ (thick), and CdS/IVO (thin), respectively. In particular, the change of work functions between CdS/IVO:Eu³⁺ (thin) and CdS/IVO (thin) heterojunctions directly affects energy transfer from CdS to IVO

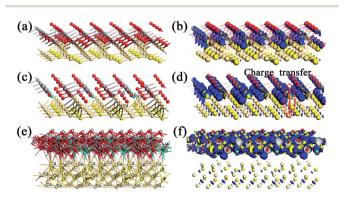


Fig. 8 The geometric structures and charge density difference of (a and b) CdS/IVO (thin), (c and d) CdS/IVO: Eu^{3+} (thin), and (e and f) CdS/IVO: Eu^{3+} (thick).

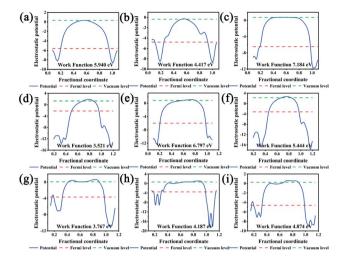


Fig. 9 Calculated Fermi levels, vacuum levels, and work functions of the (112) surface of samples with different section thicknesses and vacuum layer thicknesses: (a) CdS (thin), (b) CdS (thick), (c) IVO (thin), (d) IVO (thick), (e) IVO–Eu (thin), (f) IVO–Eu (thick), (g) CdS/IVO:Eu³⁺ (thin), (h) CdS/IVO:Eu³⁺ (thick), and (i) CdS/IVO (thin).

in the CdS/IVO:Eu³⁺ heterojunction, which is conducive to the improvement of interfacial charge transfer and photocatalytic performance.

Fig. 10 shows the calculated band structure and density of states of the (112) surface for IVO (thin), IVO–Eu (thin), CdS (thin), CdS/IVO (thin), CdS/IVO:Eu³⁺ (thin), and CdS/IVO:Eu³⁺ (thick). The results show that the band gap of IVO becomes narrower after Eu doping. In addition, the band structure and density of states are also related to the section thickness and vacuum layer thickness. Therefore, different crystal planes and atoms of the material can be exposed by controlling the synthesis conditions, so as to enhance the photocatalytic performance.

Fig. S12[†] shows the optical properties of bulk CdS, IVO, and IVO:Eu³⁺ obtained by theoretical simulation. Fig. S13[†] shows

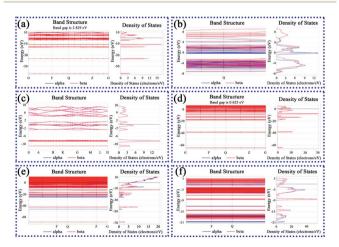


Fig. 10 Band structure and density of states for the (112) surface: (a) IVO (thin), (b) IVO-Eu (thin), (c) CdS (thin), (d) CdS/IVO (thin), (e) CdS/ IVO:Eu³⁺ (thin), and (f) CdS/IVO:Eu³⁺ (thick).

the theoretical optical properties of the (112) surface of CdS, IVO, IVO–Eu, CdS/IVO, CdS/IVO:Eu³⁺ with different section thicknesses and vacuum layer thicknesses. It is noted that the dielectric function is related to the electronic transition and electronic structure in the crystal and can reflect the band structure and optical properties of the materials. The imaginary part of the dielectric function describes the actual transition between the occupied state and the unoccupied state. The results show that we can reasonably design the composite structures by designing the particle size, the exposed crystal surface, and the proportion of composite materials, so as to meet our different needs.

4. Conclusions

Starting from the controllable construction at the atomic and molecular scales, and considering the special structure of Eu³⁺ ions and the advantages of CdS/IVO:Eu³⁺ Type-II heterojunctions, we propose that the single-atom Eu³⁺ can not only modify IVO by doping, but can also act as an interface charge transfer bridge between IVO and CdS, thereby fully promoting the charge separation between the interfaces and improving the photocatalytic activity of CO2 reduction. As expected, small CdS particles are compounded on the InVO₄ plate with a uniform thickness through the combination of a hydrothermal process and in situ synthesis method. Under the irradiation of visible light, the yield of CH₄ for 10-CdS/IVO:Eu³⁺ reached 8.24 µmol g^{-1} h⁻¹, which was 5.3 times and 10.6 times that of IVO and CdS, and the CO yield was 11.02 μ mol g⁻¹ h⁻¹, which was almost 2.4 times that of IVO and 1.9 times that of CdS, respectively. The XPS test results show that the In 3d, O 1s, V 2p, and Eu 3d peaks of 10-CdS/IVO:Eu³⁺ move slightly towards the low energy direction, indicating that the charge density around In, O, V, and Eu increases. The positions of Cd 3d and S 2p peaks are both shifted to the high binding energy direction, which indicates that the charge density around Cd and S decreases. Thus, the Eu³⁺ ions effectively promote the charge transfer between CdS and IVO:Eu³⁺, and electrons can transfer from CdS to IVO. Through a series of DFT calculations, the results of the band structure, work function and CO₂ adsorption energy successfully proved that the Eu³⁺ single atom has a dual function of CO₂ activation and charge separation promotion. In addition, the combined mode of Eu³⁺ single atom catalysis and the heterojunction makes full use of their advantages. It is expected that the results of this work can provide more theoretical and experimental design support ideas and prospects for RE single atom catalysts, and open up a novel pathway for the design of photocatalytic systems with high catalytic performance.

Author contributions

Y. L. analyzed the experimental data and wrote the original draft. Y. L. performed major experiments. G. W. designed the concept of this work. G. W. performed the computational study. G. W. and Y. Q. revised the manuscript. All the authors discussed the results and commented on the manuscript.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 D. Wang and Y. Li, Adv. Mater., 2011, 23, 1044-1060.
- 2 S. Ji, Y. Qu, T. Wang, Y. Chen, G. Wang, X. Li, J. Dong, Q. Chen, W. Zhang, Z. Zhang, S. Liang, R. Yu, Y. Wang, D. Wang and Y. Li, *Angew. Chem., Int. Ed.*, 2020, **59**, 10651– 10657.
- 3 Y. Wang, D. Wang and Y. Li, Adv. Mater., 2021, 33, 2008151.
- 4 Y. Wang, S. Wang and X. Lou, *Angew. Chem., Int. Ed.*, 2019, 58, 17236–17240.
- 5 Q. Shi, Z. Li, L. Chen, X. Zhang, W. Han, M. Xie, J. Yang and L. Jing, *Appl. Catal., B*, 2019, **244**, 641–649.
- 6 W. Liu, X. Li, C. Wang, H. Pan, W. Liu, K. Wang, Q. Zeng,
 R. Wang and J. Jiang, *J. Am. Chem. Soc.*, 2019, 141, 17431– 17440.
- 7 N. Zhang, C. Ye, H. Yan, L. Li, H. He, D. Wang and Y. Li, *Nano Res.*, 2020, **13**, 3165–3182.
- 8 J. Ran, M. Jaroniec and S. Qiao, Adv. Mater., 2018, 30, 1704649.
- 9 S. Cao, B. Shen, T. Tong, J. Fu and J. Yu, *Adv. Funct. Mater.*, 2018, **28**, 1800136.
- 10 S. Ji, Y. Chen, X. Wang, Z. Zhang, D. Wang and Y. Li, *Chem. Rev.*, 2020, **120**, 11900–11955.
- 11 J. Wang, W. Liu, G. Luo, Z. Li, C. Zhao, H. Zhang, M. Zhu, Q. Xu, X. Wang, C. Zhao, Y. Qu, Z. Yang, T. Yao, Y. Li, Y. Lin, Y. Wu and Y. Li, *Energy Environ. Sci.*, 2018, **11**, 3375–3379.
- 12 J. Bian, Y. Qu, X. Zhang, N. Sun, D. Tang and L. Jing, *J. Mater. Chem. A*, 2018, **6**, 11838–11845.
- 13 J. Yang, J. Hao, S. Xu, Q. Wang, J. Dai, A. Zhang and X. Pang, *ACS Appl. Mater. Interfaces*, 2019, **11**, 32025–32037.
- 14 W. Zhao, Y. Feng, H. Huang, P. Zhou, J. Li, L. Zhang, B. Dai, J. Xu, F. Zhu, N. Sheng and D. Leung, *Appl. Catal.*, *B*, 2019, 245, 448–458.
- 15 Q. Bi, J. Wang, J. Lv, J. Wang, W. Zhang and T. Lu, *ACS Catal.*, 2018, **8**, 11815–11821.
- 16 D. Wang and Y. Li, J. Am. Chem. Soc., 2010, 132, 6280.
- 17 H. Yu, E. Haviv and R. Neumann, Angew. Chem., Int. Ed., 2020, 59, 6219–6223.
- 18 X. Li, H. Rong, J. Zhang, D. Wang and Y. Li, *Nano Res.*, 2020, 13, 1842–1855.
- 19 J. Wang, W. Liu, G. Luo, Z. Li, C. Zhao, H. Zhang, M. Zhu, Q. Xu, X. Wang, C. Zhao, Y. Qu, Z. Yang, T. Yao, Y. Li, Y. Lin, Y. Wu and Y. Li, *Energy Environ. Sci.*, 2018, **11**, 3375–3379.

- 20 J. Zhang, S. Wu, X. Lu, P. Wu and J. Liu, *ACS Nano*, 2019, **13**, 14152–14161.
- 21 R. Kuriki, M. Yamamoto, K. Higuchi, Y. Yamamoto, M. Akatsuka, D. Lu, S. Yagi, T. Yoshida, O. Ishitani and K. Maeda, *Angew. Chem.*, *Int. Ed.*, 2017, 56, 4867–4871.
- 22 G. Zhao, W. Zhou, Y. Sun, X. Wang, H. Liu, X. Meng,
 K. Chang and J. Ye, *Appl. Catal.*, B, 2018, 226, 252–257.
- 23 H. Xu, Y. Wang, X. Dong, N. Zheng, H. Ma and X. Zhang, *Appl. Catal.*, *B*, 2019, **257**, 117932.
- 24 Q. Han, X. Bai, Z. Man, H. He, L. Li, J. Hu, A. Alsaedi, T. Hayat, Z. Yu, W. Zhang, J. Wang, Y. Zhou and Z. Zou, *J. Am. Chem. Soc.*, 2019, **141**, 4209–4213.
- 25 S. He, Z. Yang, X. Cui, X. Zhang and X. Niu, *Chem*, 2020, **260**, 127548.
- 26 R. Kuriki, H. Matsunaga, T. Nakashima, K. Wada, A. Yamakata, O. Ishitani and K. Maeda, J. Am. Chem. Soc., 2016, 138, 5159–5170.
- 27 R. Qi, P. Yu, J. Zhang, W. Guo, Y. He, H. Hojo, H. Einaga, Q. Zhang, X. Liu, Z. Jiang and W. Shangguan, *Appl. Catal.*, *B*, 2020, 274, 119099.
- 28 Z. Shen, H. Li, H. Hao, Z. Chen, H. Hou, G. Zhang, J. Bi,
 S. Yan, G. Liu and W. Gao, *J. Photochem. Photobiol.*, *A*, 2019, 380, 111864.
- 29 M. Kuehnel, K. Orchard, K. Dalle and E. Reisner, J. Am. Chem. Soc., 2017, 139, 7217-7223.
- 30 S. Wang, M. Xu, T. Peng, C. Zhang, T. Li, I. Hussain, J. Wang and B. Tan, *Nat. Commun.*, 2019, **10**, 10.
- 31 C. Gao, J. Low, R. Long, T. Kong, J. Zhu and Y. Xiong, *Chem. Rev.*, 2020, **120**, 12175–12216.
- 32 Y. Zhao, Z. Han, G. Gao, W. Zhang, Y. Qu, H. Zhu, P. Zhu and G. Wang, *Adv. Funct. Mater.*, 2021, **31**, 2104976.
- 33 J. Shan, F. Raziq, M. Humayun, W. Zhou, Y. Qu, G. Wang and Y. Li, *Appl. Catal.*, *B*, 2017, **219**, 10–17.
- 34 S. Bai, C. Gao, J. Low and Y. Xiong, *Nano Res.*, 2019, **12**, 2031–2054.
- 35 Q. Chen, G. Gao, Y. Zhang, Y. li, H. Zhu, P. Zhu, Y. Qu,
 G. Wang and W. Qin, *J. Mater. Chem. A*, 2021, 9, 15820– 15826.

- 36 Z. Han, Y. Zhao, G. Gao, W. Zhang, Y. Qu, H. Zhu, P. Zhu and G. Wang, *Small*, 2021, 17, 2101089.
- 37 X. Ye, C. Yang, X. Pan, J. Ma, Y. Zhang, Y. Ren, X. Liu, L. Li and Y. Huang, *J. Am. Chem. Soc.*, 2020, **142**, 19001–19005.
- 38 L. Wang, W. Chen, D. Zhang, Y. Du, R. Amal, S. Qiao, J. Bf and Z. Yin, *Chem. Soc. Rev.*, 2019, 48, 5310–5349.
- 39 J. Liu, X. Kong, L. Zheng, X. Guo, X. Liu and J. Shui, ACS Nano, 2020, 14, 1093–1101.
- 40 K. Choi, D. Kim, B. Rungtaweevoranit, C. Trickett, J. Barmanbek, A. Alshammari, P. Yang and O. Yaghi, *J. Am. Chem. Soc.*, 2017, **139**, 356–362.
- 41 S. Bao, H. Yu, G. Gao, H. Zhu, D. Wang, P. Zhu and G. Wang, 2021, DOI: 10.1007/s12274-021-3886-x.
- 42 S. Wang, B. Zhu, M. Liu, L. Zhang, J. Yu and M. Zhou, *Appl. Catal.*, *B*, 2019, **243**, 19–26.
- 43 J. Yang, W. Li, D. Wang and Y. Li, Small Struct., 2021, 2, 2000051.
- 44 Z. Wang, C. Li and K. Domen, *Chem. Soc. Rev.*, 2019, 48, 2109–2125.
- 45 Z. Zhuang, Q. Kang, D. Wang and Y. Li, *Nano Res.*, 2020, **13**, 1856–1866.
- 46 Y. Wang, J. Mao, X. Meng, L. Yu, D. Deng and X. Bao, *Chem. Rev.*, 2019, **119**, 1806–1854.
- 47 G. Wang, C. He, R. Huang, J. Mao, D. Wang and Y. Li, *J. Am. Chem. Soc.*, 2020, **142**, 19339–19345.
- 48 N. Wetchakun, P. Wanwaen, S. Phanichphant and K. Wetchakun, *RSC Adv.*, 2017, 7, 13911–13918.
- 49 X. Wu, Y. Li, G. Zhang, H. Chen, J. Li, K. Wang, Y. Pan, Y. Zhao, Y. Sun and Y. Xie, *J. Am. Chem. Soc.*, 2019, 141, 5267–5274.
- 50 X. Yuan, L. Jiang, J. Liang, Y. Pan, J. Zhang, H. Wang, L. Leng,
 Z. Wu, R. Guan and G. Zeng, *Chem. Eng. J.*, 2019, 356, 371–381.
- 51 C. Chen, T. Wu, H. Wu, H. Liu, Q. Qian, Z. Liu, G. Yang and B. Han, *Chem. Sci.*, 2018, 9, 8890–8894.
- 52 Z. Wang, X. Zheng and D. Wang, *Nano Res.*, 2021, DOI: 10.1007/s12274-021-3794-0.