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## Sulfide-mediated growth of NIR luminescent Pd/Ag atomically precise nanoclusters†

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An essential feature of coinage metal nanoclusters (NCs) is their photoluminescence (PL), which spans a wide range of wavelengths from visible to near-infrared regions (NIR-I/II). A key challenge for synthetic chemists is to develop materials capable of efficient spectral change with maximum efficiency. Herein, we report novel dithiolate-protected bimetallic Pd-Ag NCs of the type  $[\text{PdAg}_{16}\text{S}_2\{\text{S}_2\text{P}(\text{OR})_2\}_{12}]$  ( $\text{R} = {}^i\text{Pr}$ , **1<sub>Pr</sub>** and **1<sub>Bu</sub>**) and  $[\text{Pd}_6\text{Ag}_{14}\text{S}\{\text{S}_2\text{P}(\text{O}^i\text{Bu})_2\}_{12}]$  (**2<sub>Bu</sub>**). Sulfide-mediated expansions of NCs result in unique PL in the NIR-I region for **1<sub>Pr</sub>** and **1<sub>Bu</sub>** ( $\lambda_{\text{max}} = 808$  and  $811$  nm) and the NIR-II region for **2<sub>Pr</sub>** ( $\lambda_{\text{max}} = 1007$ ) at 77 K. NIR PL enhancement largely depends on structural modification with the sulfide anions at the central position. DFT calculations indicate that the PL properties are associated with  $4d_x(\text{Pd})/3p_x(\text{S}) \rightarrow 5s/5p(\text{Ag})$  excitation, resulting from the existence of S–Pd(0)–S motifs in both **1** and **2**. The electrochemical gaps of **1<sub>Pr</sub>**, **1<sub>Bu</sub>**, and **2<sub>Bu</sub>** are recorded by SWV.

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### Introduction

Atomically precise nanoclusters (NCs) have emerged as a material of the future, offering unique advantages over their bulk counterparts. One key improvement is the ability to modulate the energy levels and tuning energy band gaps, thus influencing both optical and electrochemical performances. Photoluminescence (PL) in the near-infrared (NIR) region is highly desirable due to its minimal interaction with biological tissues, making it a safe and effective tool for various applications in the biomedical field,<sup>1,2</sup> such as cell labeling,<sup>3</sup> bio-sensing,<sup>4</sup> and bioimaging.<sup>5–7</sup>

Over the past decade, substantial progress has been made in synthesizing and characterizing coinage metal NCs with enhanced PL quantum yields (PLQYs). However, achieving PL in the NIR-II region (1000–1700 nm) remains challenging. Coinage metal NCs have shown some promise, with a few examples of gold,<sup>8–15</sup> silver<sup>16</sup> and copper<sup>17</sup> reported. While noble metal-doped or alloyed Ag NCs can extend emission from the UV/vis to the NIR-I range,<sup>18–31,38–40</sup> achieving longer

wavelengths (1000–1700 nm) remains elusive. To achieve the NIR-II photoluminescence of silver NCs, we propose in this report the addition of a sulfur heteroatom into the metallic framework to induce emission at longer wavelengths. Various NC morphologies have been isolated due to the versatility of sulfide, which can occupy multiple positions within the NC, not just the central one.<sup>32,33</sup> In several cases, the inclusion of sulfide in the NC primary sphere directs the growth of the subsequent layers and plays a crucial role in the assembly of large arrangements.<sup>34–37</sup>

Herein, we report the synthesis, composition, and optical characteristics of the selective incorporation of sulfide anions into positions of the metallic primary layer, resulting in bimetallic Pd–Ag NCs  $[\text{PdAg}_{16}\text{S}_2\{\text{S}_2\text{P}(\text{OR})_2\}_{12}]$  ( $\text{R} = {}^i\text{Pr}$ , **1<sub>Pr</sub>** and **1<sub>Bu</sub>**) and  $[\text{Pd}_6\text{Ag}_{14}\text{S}\{\text{S}_2\text{P}(\text{O}^i\text{Bu})_2\}_{12}]$  (**2<sub>Bu</sub>**). Single-crystal X-ray diffraction (SCXRD), X-ray photoelectron spectroscopy (XPS), ESI-MS, optical analysis, and multinuclear NMR spectroscopy determined the composition and structural properties and their electronic structures were analyzed through DFT calculations.

### Results and discussion

Building upon our earlier report on sulfide-centered Pd/Ag of the type  $[\text{Pd}_6\text{Ag}_{14}(\text{S})\{\text{S}_2\text{P}(\text{O}^n\text{Pr})_2\}_{12}]$  (**2<sub>Pr</sub>**), which involves the cleavage of the P–S bond from the dithiophosphate and the *in situ* sulfide generation,<sup>19</sup> we sought to develop a more general protocol for incorporating sulfide into NCs and evaluate its reproducibility. To achieve this, we varied the alkoxy

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substituent on the dithiophosphate ligand (dtp), utilizing both isopropoxy and isobutoxy derivatives.

This strategy (Scheme 1) successfully yielded two novel disulfide nanoclusters,  $[\text{PdAg}_{16}(\text{S})_2\{\text{S}_2\text{P}(\text{OR})_2\}_{12}]$  ( $\mathbf{1}_{\text{Pr}}$ ,  $\text{R} = \text{Pr}$ ;  $\mathbf{1}_{\text{Bu}}$ ,  $\text{R} = \text{Bu}$ ), in addition to the sulfide-centered clusters,  $[\text{Pd}_6\text{Ag}_{14}(\text{S})\{\text{S}_2\text{P}(\text{O}^i\text{Bu})_2\}_{12}]$  ( $\mathbf{2}_{\text{Bu}}$ ), of which the  $[\text{Pd}_6\text{Ag}_{14}(\text{S})\{\text{S}_2\text{P}(\text{O}^i\text{Pr})_2\}_{12}]$  ( $\mathbf{2}_{\text{Pr}}$ ) relative is known.<sup>19</sup> The products are isolated and purified by column chromatography, affording  $\mathbf{1}$  and  $\mathbf{2}$  in moderate yields.

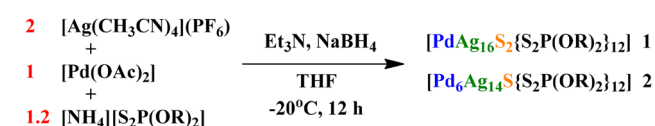
The purity and composition of the isolated NCs were confirmed by mass spectrometry, Fig. 1. The mass spectrum of  $\mathbf{1}_{\text{Pr}}$  exhibits a dominant peak assigned to  $[\mathbf{1}_{\text{Pr}} + \text{Ag}]^+$  at  $m/z$  4563.4390 Da (calc.:  $m/z$  4563.4375 Da), Fig. 1a. The simulated peak is given in Fig. S1,<sup>†</sup> compared with a silver and proton adduct to demonstrate the presence of a palladium atom. The dominant  $m/z$  peak for  $\mathbf{1}_{\text{Bu}}$  is assigned to the silver adduct  $[\mathbf{1}_{\text{Bu}} + \text{Ag}]^+$  at 4899.8438 Da (calc.:  $m/z$  4899.8140 Da), Fig. 1b and S2.<sup>†</sup> Similarly, cluster  $\mathbf{2}_{\text{Bu}}$  forms a silver adduct of the type  $[\mathbf{2}_{\text{Bu}} + \text{Ag}]^+$  at  $m/z$  5184.6678 Da (calc.:  $m/z$  5184.5506 Da), Fig. 1c. The experimental and simulated isotopic distributions

exhibit a strong correlation, further supporting the assigned compositions. Additionally, XPS analysis confirms the presence of Pd(0) and Ag(I) in the NCs, as evidenced by their characteristic binding energies, Fig. S3–5.<sup>†</sup> The temperature-dependent UV-vis spectra are shown in Fig. S13–15<sup>†</sup> and they were used to investigate the stability of the NCs.

### Structural analysis

The atomic arrangements of  $\mathbf{1}_{\text{Pr}}$  and  $\mathbf{1}_{\text{Bu}}$  were determined by SCXRD analysis and their refinement details are given in Table S1.<sup>†</sup> Selected interatomic distances are given in Table 1.  $\mathbf{1}_{\text{Pr}}$  and  $\mathbf{1}_{\text{Bu}}$  crystallize in two different space groups, with  $\mathbf{1}_{\text{Pr}}$  being solvated by a dichloromethane molecule. Otherwise, they are isostructural (apart from small differences in the ligand coordination modes, which are discussed later). Thus, the following discussion is based on  $\mathbf{1}_{\text{Bu}}$  (see Fig. 2). Its highly unique core structure features a Pd-centered  $\text{Ag}_{10}$  pentagonal antiprism kernel, with two sulfides capping each of the pentagonal faces (Fig. 2a). It can thus be seen also as a Pd-centered  $\text{Ag}_{10}\text{S}_2$  icosahedron with approximate  $D_{5d}$  symmetry and can be compared with typical species having Pd@ $\text{Ag}_{12}$  icosahedral cores.<sup>21</sup> Substituting two vertices with sulfur atoms results in a large contraction compared to the regular  $\text{Ag}_{12}$  icosahedron, with Pd...S distances (ca. 2.30 Å) much shorter than the Pd...Ag in Pd@ $\text{Ag}_{12}$  species (ca. 2.90 Å).

Furthermore, the differential bonding between Pd–S and Ag–S is crucial for stabilizing this unique kernel configuration, which is unlikely to form in the absence of the central Pd



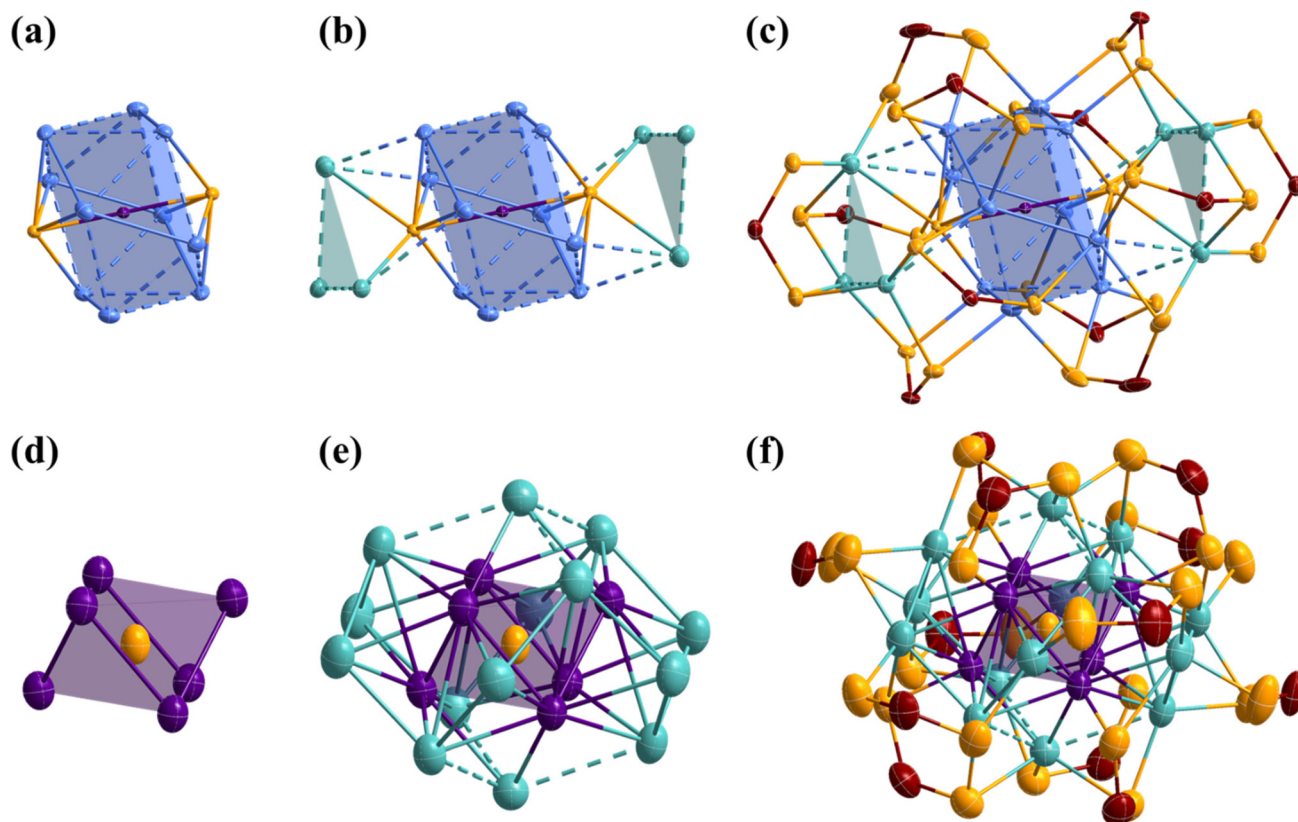
Scheme 1 Strategic incorporation of sulfide into bimetallic NCs.



Fig. 1 ESI-TOF-MS of spectra for  $\mathbf{1}_{\text{Pr}}$  (a),  $\mathbf{1}_{\text{Bu}}$  (b) and  $\mathbf{2}_{\text{Bu}}$  (c). Experimental (black) and simulated (orange).

Table 1 Selected bond distances (average, Å) for  $\mathbf{1}_{\text{Pr}}$ ,  $\mathbf{1}_{\text{Bu}}$ , and  $\mathbf{2}_{\text{Bu}}$  with X-ray (esd in brackets) and computed DFT [Wiberg indices in brackets]

	$\mathbf{1}_{\text{Pr}}$ , X-ray	$\mathbf{1}_{\text{Bu}}$ , X-ray	$\mathbf{1}$ , DFT	$\mathbf{2}_{\text{Bu}}$ , X-ray	$\mathbf{2}$ , DFT
Pd–S	2.296(8)	2.297(3)	2.336 [0.351]	2.2600(11)	2.301 [0.260]
Ag <sub>ker</sub> –S	2.7582(9)	2.718(3)	2.752 [0.132]	—	—
Ag <sub>cap</sub> –S	2.9168(9)	2.8265(3)	3.040 [0.041]	—	—
Pd–Ag	2.8863(3)	2.8950(9)	2.916 [0.066]	2.9071(16)	2.997 [0.060]
Pd–Pd	—	—	—	2.9433(14)	2.985 [0.074]
Ag <sub>ker</sub> –Ag <sub>ker</sub>	3.0431(4)	3.0478(14)	3.124 [0.032]	—	—
Ag <sub>ker</sub> –Ag <sub>cap</sub>	3.1199(4)	3.1454(14)	3.215 [0.022]	—	—
Ag <sub>cap</sub> –Ag <sub>cap</sub>	3.1758(5)	3.2196(15)	3.482 [0.010]	3.039(18)	3.131 [0.040]



**Fig. 2** The SCXRD structure of **1<sub>Bu</sub>** and **2<sub>Bu</sub>**. (a) The Pd@Ag<sub>10</sub>S<sub>2</sub> kernel of **1<sub>Bu</sub>**. (b) The two Ag<sub>3</sub> motifs capping the cluster core of **1<sub>Bu</sub>**. (c) Total structure of **1<sub>Bu</sub>** with ellipsoids drawn at 35%, H and C atoms omitted for clarity. (d) The S@Pd<sub>6</sub> core of **2<sub>Bu</sub>**. (e) The S@Pd<sub>6</sub>@Ag<sub>14</sub> framework of **2<sub>Bu</sub>**. (f) Total structure of **2<sub>Bu</sub>** with ellipsoids drawn at 35%, H and C atoms are omitted for clarity (color code: Pd, purple; Ag<sub>ker</sub>, blue; Ag<sub>cap</sub>, green; S, yellow; P, maroon).

atom. Another structural distinctive feature of **1** is the arrangement of the secondary silver layer around the Pd@Ag<sub>10</sub>S<sub>2</sub> core. The six peripheral silver atoms are arranged into two triangular Ag<sub>3</sub> capping motifs, with only two atoms per triangle in bonding contacts with the Ag<sub>10</sub>S<sub>2</sub> icosahedron (Fig. 2b). These triangular arrays are unprecedented in silver-rich NCs, where capping atoms usually occupy symmetrically favored positions dictated by the ligand shell. Consequently, the Ag<sub>cap</sub>...Ag<sub>cap</sub> distances in these peripheral motifs are notably longer than the Ag<sub>ker</sub>...Ag<sub>ker</sub> distances within the kernel. Two ligand coordination modes, six  $\eta^4$  ( $\mu_2, \mu_2$ ), and six  $\eta^3$  ( $\mu_2, \mu_1$ ), are observed in **1<sub>Pr</sub>**. In contrast, **1<sub>Bu</sub>** exhibits three coordination modes, namely two  $\eta^5$  ( $\mu_3, \mu_2$ ), four  $\eta^4$  ( $\mu_2, \mu_2$ ), and six  $\eta^3$  ( $\mu_2, \mu_1$ ). The increase in the denticity in **1<sub>Bu</sub>** occurs for two ligands bridging the triangular Ag<sub>3</sub> motif to the kernel. This is accommodated by the adaptability of the sulfide, which is  $\mu_7$  in **1<sub>Bu</sub>** compared to  $\mu_8$  in **1<sub>Pr</sub>**. Notably, two of the six  $\eta^3$  ( $\mu_2, \mu_1$ ) in both **1<sub>Pr</sub>** and **1<sub>Bu</sub>** are located at unique positions on the triangular Ag<sub>3</sub> motifs of the nanocluster, where they do not interact with the central core (Fig. 2c).

The SCXRD structure of **2<sub>Bu</sub>** is shown in Fig. 2 and selected metrical data are given in Table 1. **2<sub>Bu</sub>** is found isostructural with the previously reported **2<sub>Pr</sub>**.<sup>19</sup> It shows a central sulfur

atom directing the NC shape through its encapsulation inside a Pd<sub>6</sub>Ag<sub>2</sub> rhombohedron (distorted cube), the latter lying within an Ag<sub>6</sub> octahedron. This S@(Pd<sub>6</sub>Ag<sub>2</sub>)@Ag<sub>6</sub> framework is surrounded by six capping Ag atoms and twelve dtp ligands. The latter exhibit two coordination modes, namely six  $\eta^3$  ( $\mu_2, \mu_1$ ) and six  $\eta^4$  ( $\mu_2, \mu_2$ ) (Fig. 2f). The whole structure is of approximate *S*<sub>6</sub> symmetry. Each Pd atom in **2<sub>Bu</sub>** is coordinated to the central sulfide ion and an S atom from a dtp ligand in a nearly linear fashion (S–Pd–S = 177°). In addition to two sulfur atoms, six Ag atoms and two Pd atoms complete the coordination sphere around each Pd atom. Notably, one of the Pd–Ag contacts is particularly long (~3.2 Å). The completely different metallic frameworks in **1** and **2** underscore the structural diversity achievable through ligand manipulation and the impact of varying metal ratios within these systems.

The <sup>1</sup>H NMR data confirm the presence of the isopropyl or isobutyl group in compounds **1** and **2**, as shown in Fig. S6c, d and S7b.† The **1<sub>Pr</sub>** <sup>31</sup>P NMR spectrum shows three distinct phosphorus environments identified by resonances at 105.31, 104.35, and 100.72 ppm. In contrast, **1<sub>Bu</sub>** displays four resonances at 108.92, 106.27, 106.00, and 102.36 ppm (Fig. S6a and b†), indicating an additional coordination mode in **1<sub>Bu</sub>**, in accordance with its solid-state structure. In the case of **2<sub>Bu</sub>**, two

$^{31}\text{P}$  NMR resonances are observed at 102.63 and 100.58 ppm, in agreement with the  $S_6$  symmetry observed in the solid state being preserved in solution (Fig. S7a).<sup>†</sup>

### Optical properties

NCs, due to their small size, typically fall below the size regime where surface plasmon resonance dominates their optical properties. However, subtle changes can significantly impact the discrete energy levels of these NCs. In general, as the size of the NC increases, the absorption spectra typically shift to longer wavelengths, attributed to the quantum confinement effect, wherein the energy levels become less discrete as the NC grows. However, larger NCs often exhibit lower photo-

luminescence quantum yields due to an increased opportunity for nonradiative decay processes. The optical spectra are shown in Fig. 3, and detailed values are compared in Table 2. Compounds **1<sub>Pr</sub>** and **1<sub>Bu</sub>** have similar absorption profiles with peaks observed at 365 and 366 nm (Fig. 3a). They emit in the NIR-I region, with peaks centered at 808 and 811 nm at 77 K, and emission lifetimes of 92 and 72  $\mu\text{s}$ , respectively (Fig. 3b and S8a and b).<sup>†</sup> Compound **2<sub>Bu</sub>** features a relatively broader absorption profile covering a range of 375–630 nm (Fig. 3a). Importantly, **2<sub>Bu</sub>** exhibits unique NIR-II emission at ambient and low temperatures, with peaks centered at  $\lambda_{\text{em}} = 1034$  nm (298 K) and  $\lambda_{\text{em}} = 1004$  nm (77 K) (Fig. 3b). A change in the alkyl substituent in **2<sub>Pr</sub>**, results in peaks centered at  $\lambda_{\text{em}} =$



**Fig. 3** (a) The absorption spectra for **1<sub>Pr</sub>**, **1<sub>Bu</sub>**, and **2<sub>Bu</sub>** recorded in 2-MeTHF. (b) Emission spectra for **1<sub>Pr</sub>** ( $\lambda_{\text{em}} = 808$  nm), **1<sub>Bu</sub>** ( $\lambda_{\text{em}} = 811$  nm), **2<sub>Pr</sub>** ( $\lambda_{\text{em}} = 1007$  nm) and **2<sub>Bu</sub>** ( $\lambda_{\text{em}} = 1004$  nm) in 2-MeTHF at 77 K. (c) Comparison of  $\lambda_{\text{em}}$  for doped silver rich NC with entries labeled by the corresponding references.

**Table 2** Spectroscopic information for series 1 and 2

	NC	$\lambda_{\text{max}}$ (nm)	$\lambda_{\text{em}}$ (nm)	$\tau$ ( $\mu\text{s}$ )	$\Phi_{\text{em}}^a$ (%)	$k_o^b$ ( $\mu\text{s}^{-1}$ )	$k_r^c$ ( $\mu\text{s}^{-1}$ )	$k_{\text{nr}}^d$ ( $\mu\text{s}^{-1}$ )
298 K	<b>1<sub>Pr</sub></b>	365	—	—	—	—	—	—
	<b>1<sub>Bu</sub></b>	366	—	—	—	—	—	—
	<b>2<sub>Pr</sub></b>	368, 408, 443, 548, 623	1046	0.0437	0.025	22.88	0.0057	22.87
	<b>2<sub>Bu</sub></b>	374, 413, 446, 545, 625	1034	0.127	0.294	7.87	0.023	7.85
77 K	<b>1<sub>Pr</sub></b>	—	808	92.3	—	—	—	—
	<b>1<sub>Bu</sub></b>	—	811	71.9	—	—	—	—
	<b>2<sub>Pr</sub></b>	—	1007	21.9	9.04	0.046	0.0041	0.042
	<b>2<sub>Bu</sub></b>	—	1004	20.6	14	0.049	0.0068	0.042

<sup>a</sup>  $\Phi_{\text{em}}$  = emission quantum yield (%). <sup>b</sup>  $k_o$  = mean excited-state decay rate constant =  $1/\text{lifetime}$  ( $\tau$ ). <sup>c</sup>  $k_r$  = radiative constant =  $\Phi_{\text{em}} \times 0.01 \times k_o$ . <sup>d</sup>  $k_{\text{nr}}$  = nonradiative constant =  $k_o - k_r$ .

1046 nm (298 K) and  $\lambda_{em} = 1007$  nm (77 K). Variations in the alkyl substituents of the ligand do not significantly influence the emission profile or lifetime. However, an increase in the size of the alkyl groups leads to an increase in the PLQY. This enhancement is possibly due to the increased electron-donating ability of the larger alkyl substituents, facilitating the emission process. The PLQYs for this series of NIR emissive compounds at ambient and low temperatures are listed in Table 2. The emission lifetimes of **2<sub>Pr</sub>** and **2<sub>Bu</sub>** increase at low temperatures, consistent with a spin-forbidden transition (Fig. S9).<sup>†</sup> A comparative analysis of emission maxima for doped Ag-rich nanoclusters reveals that NC **2** exhibits a significantly red-shifted emission wavelength compared to previously reported alloys<sup>25–30</sup> and monometallic nanoclusters<sup>31</sup> (Fig. 3c).

The cyclic voltammetry (CV) and the square wave voltammetry (SWV) of compounds **1** and **2** were recorded in organic solvent (THF and CH<sub>2</sub>Cl<sub>2</sub>) with 0.1 M [<sup>n</sup>BuN][PF<sub>6</sub>] as the supporting electrolyte. The results are summarized in Table 3 and Fig. 4 and S10–12.<sup>†</sup> At low temperature (233 K), the CV of **1<sub>Pr</sub>** exhibits an irreversible reduction process ( $E_{pc}$ ) at  $-1.23$  V (vs. Fc<sup>+</sup>/Fc) and three irreversible oxidation processes ( $E_{pa}$ ) at 0.17, 0.55 and 0.766 V, respectively (Fig. S10<sup>†</sup>). The SWV measurements confirm these redox processes, with  $E_{pc} = -1.17, -1.51,$  and  $-1.86$  V and  $E_{pa} = 0.25, 0.55, 0.94,$  and  $1.38$  V at 298 K (Fig. 4a). **1<sub>Bu</sub>** displays a similar pattern in both CV and SWV at

233 and 298 K in THF (Fig. S11<sup>†</sup>). The electrochemical HOMO–LUMO gap, which is calculated to be 1.40 eV, is different between O1 and R1. The results of **2<sub>Bu</sub>** are shown at 298 and 233 K in Fig. 4b and S12.<sup>†</sup> Its calculated electrochemical HOMO–LUMO gap is 1.13 eV.

### Bonding analysis and DFT investigation

The DFT-optimized geometry carried out on a simplified model for **1**, [PdAg<sub>16</sub>S<sub>2</sub>(S<sub>2</sub>PH<sub>2</sub>)<sub>12</sub>], is in good agreement with its SCXRD structure (see Table 1 and corresponding xyz file in the ESI<sup>†</sup>). Analysis of its electronic structure supports the first-approximation view of a linear [PdS<sub>2</sub>]<sup>4–</sup> complex stabilized by a [Ag<sub>16</sub>(S<sub>2</sub>PH<sub>2</sub>)<sub>12</sub>]<sup>4+</sup> cage, of which 12 Ag<sup>+</sup> are in contact with the encapsulated [PdS<sub>2</sub>]<sup>4–</sup>. The NAO atomic charges (Table S2<sup>†</sup>) are consistent with Pd(0) and Ag(I) metal centers.<sup>19</sup> Not considering the metal–metal contacts, 14 among the 16 Ag atoms are bonded to three dtp sulfur atoms, making them locally 16-electron centers. 4 of them not in contact with Pd (labelled Ag<sub>cap</sub>) achieve a stable, nearly planar, coordination mode, whereas 10 of them in contact with Pd are somewhat pyramidalized to allow some 4d(Pd) → 5s/5p(Ag) electron transfer. Only 2 among the 16 Ag atoms are tetracoordinated, with a trigonal-planar coordination mode to three dtp sulfur atoms, to which a weaker bonding contact with one sulfide should be added. This bonding description is supported by the average Wiberg

**Table 3** The redox potentials vs. Fc<sup>+</sup>/Fc(v) and calculated band gaps (eV)

NC	$E_{pa}^a$	$E_{pc}^a$	$E_g^E, b$	$E_g^O, c$	$E_g^{PL, d}$
<b>1<sub>Pr</sub></b> (298 K)	+0.25, +0.55, +0.94, +1.38	–1.17, –1.51, –1.86	1.42	3.40	—
<b>1<sub>Pr</sub></b> (233 K)	+0.28, +0.61, +0.97	–1.35, –1.78	1.63	—	1.53
<b>1<sub>Bu</sub></b> (298 K)	+0.18, +0.41, +1.12	–1.22, –1.91	1.40	3.39	—
<b>1<sub>Bu</sub></b> (233 K)	+0.25, +0.95	–1.41, –1.89	1.66	—	1.52
<b>2<sub>Bu</sub></b> (298 K)	–0.23, +0.21, +0.78, +1.19	–1.36, –1.63, –2.20	1.13	1.98	1.20
<b>2<sub>Bu</sub></b> (233 K)	+0.01, +0.65, +1.09, +1.40	–1.52, –1.67, –2.04	1.49	—	1.24

<sup>a</sup> Values determined from SWV spectra. <sup>b</sup>  $E_g^E = E_{pa} - E_{pc}$ . <sup>c</sup>  $E_g^O = 1240/\lambda_{max}$ ,  $\lambda_{max}$  is the lowest electronic transition in absorption wavelength. <sup>d</sup>  $E_g^{PL} = 1240/\lambda_{em}$ ,  $\lambda_{em}$  is the centered PL wavelength.



**Fig. 4** Square wave voltammetry of **1<sub>Pr</sub>**, (a) and **2<sub>Bu</sub>**, (b) recorded in THF and CH<sub>2</sub>Cl<sub>2</sub>, at 298 K containing 0.1 M [<sup>n</sup>BuN][PF<sub>6</sub>] under a N<sub>2</sub> atmosphere.

bond indices (WBI) reported in Table S2,† notably the large Pd–S<sub>sulfide</sub> value (0.351), the Ag<sub>ker</sub>⋯Ag<sub>ker</sub> and Ag<sub>ker</sub>⋯Ag<sub>cap</sub> values (0.032 and 0.022, respectively) characteristic of d<sup>10</sup>⋯d<sup>10</sup> metallophilic interactions, and the somewhat larger Pd–Ag values (0.066) indicative of some weak covalent character in addition to metallophilicity. The orbital diagram of **1** is shown in Fig. S16.† It features two highest occupied orbitals derived from the two HOMOs of the hypothetical [PdS<sub>2</sub>]<sup>4-</sup>, with 4d<sub>π</sub>(Pd) contribution mixed in an antibonding way with 3p<sub>π</sub>(S). The lowest vacant orbitals are of 4s/4p(Ag) dominant character originating from the 16-electron Ag centers.

The TD-DFT-simulated UV-vis spectrum of **1** is shown in Fig. S18.† It shows a low-energy band at 502 nm, which is a combination of the HOMO → LUMO+1, HOMO–1 → L and HOMO–1 → LUMO+1 transitions, thus of 4d<sub>π</sub>(Pd)/3p<sub>π</sub>(S) → 5s/5p(Ag) nature. This low-energy peak is hardly distinguishable in the experimental spectra of **1<sub>Pr</sub>** and **1<sub>Bu</sub>** (Fig. 3a) and we believe that its computed oscillator strength is significantly overestimated in our TD-DFT calculations. The high-energy peak involves lower 4d(Ag) levels and the LUMO and LUMO+1, thus is of 4d(Ag) → 4s/4p(Ag) nature.

Although its composition and 3D architecture look fairly different from that of **1**, the bonding in **2**, whose simplified [Pd<sub>6</sub>Ag<sub>14</sub>(S)(S<sub>2</sub>PH<sub>2</sub>)<sub>12</sub>] model we previously investigated,<sup>19</sup> presents important similarities with that in **1**. Whereas **1** possesses a central linear S–Pd(0)–S motif, **2** shows six of them, sharing the central sulfide and making a flattened octahedron. Discarding the hypercoordinated nature of the sulfide, the six S–Pd(0)–S motifs are stable 14-electron centers, while the silver atoms are all tricoordinated 16-electron centers, of which those in contact with Pd are involved in (moderate) 4d(Pd) → 5s/5p(Ag) electron transfer. As in **1**, the whole architecture is maintained by the bidentate dtp ligands. The orbital diagram of **2** (Fig. S17†) shows highest occupied levels which can be roughly described as combinations of the two π-type HOMOs of the six S–Pd(0)–S motifs of 4d<sub>π</sub>(Pd) and 3p<sub>π</sub>(S) character, as in (**1**). Because of the existence of Pd⋯Pd contacts (Table 1), the 2 × 6 occupied combinations develop a “band” according to their (rather weak) amount of Pd⋯Pd bonding/antibonding character, of which the highest are (moderately) Pd⋯Pd antibonding, thus tending to limit the width of the HOMO–LUMO separation.

The TD-DFT-simulated UV-vis spectra of **1** and **2** (Fig. S18†) are in a good agreement with their experimental counterpart (Fig. 3a). They show two major bands which are of similar nature. Their major high-energy band is of 4d(Ag/Pd) → ligand character. The oscillator strength of lowest energy bands (of 4d<sub>π</sub>(Pd)/3p<sub>π</sub>(S) → 5s/5p(Ag) nature) are overestimated with respect to experiment. In fact, in the case of **1**, this computed low-energy band (502 nm, Fig. S18†) can be associated with transitions buried in the tails observed around 400–500 nm in the **1<sub>Pr</sub>** and **1<sub>Bu</sub>** experimental spectra (Fig. 3a). In the case of **2**, this band, also of 4d<sub>π</sub>(Pd)/3p<sub>π</sub>(S) → 5s/5p(Ag) nature, results from a transition from the degenerate HOMO–1/HOMO–2 e<sub>u</sub> level to the degenerate lowest unoccupied e<sub>g</sub> level (LUMO/LUMO+1). Although of similar nature, the low-energy band in **2** is red-shifted as compared to that computed for **1**. We suggest that this

is the same transition which is responsible for the NIR-II phosphorescence of **2**. The large Stokes shift experimentally observed can be rationalized from the fact that the e<sub>u</sub> → e<sub>g</sub> transition in **2** corresponds to the depopulation of a Pd⋯Pd antibonding level and the occupation of an Ag⋯Ag bonding level, with the consequence of a non-negligible stabilization of the corresponding excited state, a situation which does not exist in **1**.

## Conclusion

This study has successfully demonstrated the strategic synthesis of sulfide-doped nanoclusters (NCs) by modulating the reaction pH with triethylamine, a method that effectively generates sulfide ions from dithiophosphate ligands. Notably, the isolation of **1<sub>Pr</sub>** and **1<sub>Bu</sub>**, which exhibit *the first silver-rich icosahedron containing two sulfur heteroatoms at its vertices*, resulted in the appearance of photoluminescence in the NIR-I region. Furthermore, the sulfide-centered NC **2<sub>Bu</sub>** exhibits remarkable NIR-II photoluminescence at both ambient and low temperatures, highlighting the potential of this synthetic strategy for producing rare NIR emissive nanoclusters. Given that the NIR-II emission originates from a PdS<sub>2</sub> → Ag transition, the targeted incorporation of additional Pd and S atoms into silver NCs is a promising strategy for further enhancing their optical properties. This study sheds light on the intricate relationship between size, structure, and optical properties in NCs, emphasizing the importance of precise control over these parameters. This finding and our controlled synthesis and ligand modification capabilities open exciting avenues for designing new materials with tailored optical responses. These advancements hold promise for applications in bioimaging, sensing, optoelectronics, and beyond.

## Author contributions

Y.-R. N., T.-H. C., and M. N. P.: investigation, data curation, formal analysis, methodology, and writing. J. Y. C., Y.-J. C., H. L., and S. K.: data curation and formal analysis. J.-Y. S.: data curation, formal analysis, and writing. C. W. L.: supervision, writing, conceptualization, project administration, and resources.

## Data availability

Crystallographic data for [**1<sub>Pr</sub>**, **1<sub>Bu</sub>**, **2<sub>Bu</sub>**] have been deposited at the Cambridge Crystallographic Data Centre under [CCDC 2379720–2379722].† Computational details (including coordinate file), X-ray structure analyses, NMR spectra, electrochemical data, ESI-MS data, and luminescence decay curves are available in the ESI.†

## Conflicts of interest

There are no conflicts to declare.

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