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# Nonlinear optical properties of metal nanoparticles: a review

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Metal nanoparticles (MNPs) hold great technological promise because of the possibility of engineering their electronic and optical properties through material design. One of the effective methods to fabricate MNPs is ion implantation. In this review, recent results on the nonlinear optical properties of nanoparticles (including copper, silver, gold, and bismuth nanoparticles) doped in various bases have been discussed. Some specific optical nonlinear properties, such as nonlinear refraction, two-photon absorption, and optical limiting, for femtosecond, picosecond, and nanosecond laser pulses have also been covered. In addition to ion implantation, we have summarized several other methods for the preparation of composite materials, and Z-scan has been used to study the nonlinear optical properties of these materials.

## 1 Nonlinear optical properties

### 1.1 Introduction

**1.1.1 Theory of nonlinear optics.** Nonlinear optics is one of the sub-disciplines of modern optics, with an important academic status on a par with laser physics. Nonlinear optics studies involve various kinds of nonlinear effects that occur in the interaction between laser and matter.<sup>1,2</sup> The research object of nonlinear optics mainly involves the new phenomena and new effects in the interaction process of strong laser radiation and materials, including an in-depth understanding of the causes and the process regularity and exploration of their possible applications in the current or future development of disciplines. The interaction of a laser with a nonlinear optical material causes a modification of the optical properties of the material system, such that the next photon that arrives realizes a different material.<sup>3</sup> Typically, only laser light is sufficiently intense to generate the nonlinear optical phenomena; therefore, the beginning of this research field is often considered to be since the discovery of second-harmonic generation by Franken *et al.* in 1961,<sup>4</sup> one year after the construction of the first laser by Maiman.<sup>5</sup> Although the observation of most nonlinear optical phenomena required laser radiation, some classes of nonlinear optical effects were known long before the invention of the laser. The most prominent examples of these phenomena include Pockel's and Kerr electro-optic effects.<sup>6</sup> The theory of nonlinear optics is based on the well-understood theory of linear optics, particularly on the part covering the interaction of light and matter.

Some important formulas describing ordinary optical phenomena often exhibit mathematical linear characteristics. For example, to explain the refractive index, scattering, and birefringence of the medium, an important physical mass-dielectric polarization vector  $P$  was introduced, and it was speculated that it had a simple linear relationship with the incident wave field  $E$ , that is

$$P = \chi E \quad (1)$$

where the coefficient  $\chi$  is the electromagnetism of the medium.

To make a perfect interpretation of the effects of optical secondary harmonics, third harmonic and rectification effects were observed in a series of media. In this regard, it is necessary to speculate that under the action of an intense laser, the dielectric strength of the medium is no longer linearly related to the intensity of the incident light field, but rather a more general power relationship (including the nonlinear component, which is proportional to the high power of the field) must be applied. That is

$$P = \chi^{(1)}E + \chi^{(2)}EE + \chi^{(3)}EEE + \dots \quad (2)$$

where  $\chi^{(1)}$ ,  $\chi^{(2)}$ , and  $\chi^{(3)}$  are the primary (linear), quadratic (nonlinear), and cubic (nonlinear) polarizabilities of the medium, respectively, and in general, they are the coefficients of the tensor form. By substituting the abovementioned expression of the electrode polarization into the Maxwell equation and utilizing a set of nonlinear electromagnetic wave equations with high order wave field strength, multiplier radiation can be generated when a single frequency of light is incident on a particular medium; when a variety of different frequencies of light are simultaneously incident, they can couple with each other through the media; then, a mixed radiation is generated at the new frequency. From the viewpoint of this new nonlinear

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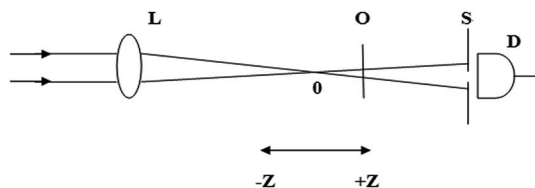


Fig. 1 Schematic of the basic device of the Z-scan method.



Fig. 2 Z-Scan curve for (left) negative optical nonlinear samples and (right) positive optical nonlinear samples.

a measurement material. In 1992, Sheik-Bahae *et al.* proposed a two-color Z-scan method and applied it to measure non-degenerate nonlinear and non-degenerate nonlinear two-photon absorption coefficients. Within a short time period, the Z-scan technology has continuously improved, and it has been tested and studied in other aspects such as thermal lens effects, laser beam quality testing, and Gaussian beam transmission characteristics. Z-Scanning technology has since developed into having a significant practical value among the test methods used in the study of nonlinear optical properties.

**1.2.1 The basic device of the Z-scanning method.** The basic device of the Z-scanning method and its optical path are shown in Fig. 1, where the drift of a Gaussian beam convergence output travels through the lens L and the smallest section of the beam crosses at the 0 point of the Z axis; O stand for the test object, placed near the 0 of the Z axis, S is the limiting aperture, and D is the photodetector. During the test, the object to be tested moves along the Z axis and records the relationship between the light intensity and Z value changes. The Z-scanning method of the recording device and the test process are not complicated, but this method provides very rich information about the tested object (Fig. 2).

**1.2.2 Measurement of the nonlinear refractive index.** The power of the incident light and the linear transmittance of the orifice can, respectively, be described as follows:

$$P_i(t) = \pi\omega_0^2 I_0(t)/2 \quad (3)$$

$$s = 1 - \exp(-2r_a^2/\omega_a^2) \quad (4)$$

where  $\omega_0$  is the waist radius of the beam,  $\omega_a$  is the radius of the screen,  $r_a$  is the hole radius,  $I_0(t)$  is the intensity of the light at the focal point, and the normalized transmittance is given by

$$T(z) = \frac{\int_{-\infty}^{+\infty} P_i[\Delta\phi_0(t)]dt}{s \int_{-\infty}^{+\infty} P_i(t)dt} \quad (5)$$

The phase change of the wave surface at the focal point on the axis is

$$\Delta\phi_0 = k\Delta n_0(t)L_{\text{eff}} \quad (6)$$

Among them,  $k = \frac{2\pi}{\lambda}$ ,  $\lambda$  is the laser wavelength,  $L_{\text{eff}} = (1 - e^{-\alpha L})/\alpha$ ,  $\alpha$  is the linear absorption coefficient,  $L$  is the sample thickness, and  $\Delta n_0 = \gamma I_0(t)$ .

By definition  $\Delta T_{P-V} = T_P - T$ , which represents the difference in the normalized transmittance of the peak-valley in the normalized Z-scan curve, when  $\Delta\phi_0 \leq \pi$

$$\Delta T_{P-V} \approx 0.406(1 - s^{0.25})|\Delta\phi_0| \quad (7)$$

Therefore, according to the obtained Z-scan curve,  $\gamma$  can be obtained using (6) and (7).

**1.2.3 Measurement of the coefficient of nonlinear absorption.** Optical nonlinear materials tend to have significant nonlinear absorption, and the absorption coefficient of the material can be written as

$$\alpha(I) = \alpha + \beta I \quad (8)$$

where  $\alpha$  is the linear absorption coefficient and  $\beta$  is the nonlinear absorption coefficient. Then, the normalized transmittance is given by<sup>34</sup>

$$T(z, s = 1) = \frac{1}{\sqrt{\pi}q_0(z, 0)} \int_{-\infty}^{+\infty} \ln[1 + q_0(z, 0)e^{-\tau}]d\tau \quad (9)$$

$$q_0(z, t) = \beta I_0(t)L_{\text{eff}}/(1 + z^2/z_0^2), \text{ when } |q_0| < 1,$$

$$T(z, s = 1) = \sum_{m=0}^{\infty} \frac{[-q_0(z, 0)]^m}{(m+1)^{3/2}} \quad (10)$$

When a Z-scan is performed without an aperture, numerical fitting can be conducted on the experimental results according to the abovementioned formula.

## 2 NOPs of metal ions implanted into various matrices

In many studies, composite materials with MNPs exhibiting nonlinear optical characteristics were fabricated by various methods and then generally studied using lasers operating at frequencies corresponding to the spectral range of the SPR, as shown in Table 1. However, in fact, when novel materials with MNPs are used, their properties should be systematically studied. Hence, to create new materials promising for practical use in laser systems and integral optics and to optimize their characteristics, the nonlinear optical properties of these materials should be studied not only in the SPR spectral region but also at the specific lasers for industry need.





Table 1 Types of optically transparent dielectric and semiconductor matrices with metal nanoparticles

Metal (ion)	Matrix	Synthesis conditions: energy ( $E$ ), keV, dose ( $D$ ), ion per $\text{cm}^2$ , current density ( $J$ ), $\mu\text{A cm}^{-2}$ , annealing temperature ( $T$ ), $^\circ\text{C}$ , time ( $t$ ) ps	Study	Laser parameters: wavelength ( $\lambda$ ), nm, pulse duration ( $\tau$ ), ps, repetition rate ( $\nu$ ), Hz, intensity ( $I_0$ ), $\text{W cm}^{-2}$ , pulse energy ( $P$ ) mJ	Nonlinear parameters: refract. coeff. ( $n_2$ ), $\text{cm}^2 \text{W}^{-1}$ , absorption coeff. ( $\beta$ ), $\text{cm W}^{-1}$ , satur. intensity ( $I_{\text{sat}}$ ), $\text{W cm}^{-2}$ , $\text{Re}[\chi^{(3)}]$ , $\text{Im}[\chi^{(3)}]$ , $[\chi^{(3)}]$ , esu	Author
Cu	STO ( $\text{SrTiO}_3$ )	$D = 1 \times 10^{16}$ , $t < 1$	DFWM, Z-scan	$\lambda = 775 \text{ nm}$ , $\tau = 250 \text{ fs}$ , $\nu = 1000 \text{ Hz}$	$\beta = 1.78 \times 10^{-12}$ , $\chi^{(3)} = 1.55 \times 10^{-10}$ , $[\chi^{(3)}] = 2.2 \times 10^{-10}$	Cetin <i>et al.</i> 2010 (ref. 35)
Cu	STO ( $\text{SrTiO}_3$ )	$D = 5 \times 10^{16}$ , $t = 2.46$	DFWM, Z-scan	$\lambda = 775 \text{ nm}$ , $\tau = 250 \text{ fs}$ , $\nu = 1000 \text{ Hz}$	$\beta = 4.23 \times 10^{-12}$ , $\chi^{(3)} = 3.66 \times 10^{-10}$ , $[\chi^{(3)}] = 4.5 \times 10^{-10}$	
Tb	STO ( $\text{SrTiO}_3$ )	$D = 1 \times 10^{17}$ , $t = 3.58$	DFWM, Z-scan	$\lambda = 775 \text{ nm}$ , $\tau = 280 \text{ fs}$ , $\nu = 1000 \text{ Hz}$	$\beta = 6.15 \times 10^{-12}$ , $\chi^{(3)} = 5.33 \times 10^{-10}$ , $[\chi^{(3)}] = 6.54 \times 10^{-10}$	
Tb	STO ( $\text{SrTiO}_3$ )	$D = 1 \times 10^{17}$ , $t = 3.58$	DFWM, Z-scan	$\lambda = 775 \text{ nm}$ , $\tau = 280 \text{ fs}$ , $\nu = 1000 \text{ Hz}$	$\beta = 5.53 \times 10^{-12}$ , $\chi^{(3)} = 4.79 \times 10^{-10}$ , $[\chi^{(3)}] = 5.9 \times 10^{-10}$	
Tb	STO ( $\text{SrTiO}_3$ )	$D = 5 \times 10^{16}$ , $t = 3.55$	DFWM, Z-scan	$\lambda = 775 \text{ nm}$ , $\tau = 280 \text{ fs}$ , $\nu = 1000 \text{ Hz}$	$\beta = 11.6 \times 10^{-12}$ , $\chi^{(3)} = 10 \times 10^{-10}$ , $[\chi^{(3)}] = 12.5 \times 10^{-10}$	
Tb	STO ( $\text{SrTiO}_3$ )	$D = 1 \times 10^{17}$ , $t = 4.15$	DFWM, Z-scan	$\lambda = 775 \text{ nm}$ , $\tau = 280 \text{ fs}$ , $\nu = 1000 \text{ Hz}$	$\beta = 13.4 \times 10^{-12}$ , $\chi^{(3)} = 11.6 \times 10^{-10}$ , $[\chi^{(3)}] = 14.2 \times 10^{-10}$	
As <sup>2+</sup>	GaAs-AlGaAs	$E = 4 \times 10^3$ , $D = 5 \times 10^{12}$ , $T = 775$ , $t > 60$		$\lambda = 1545\text{--}1565$ , $\nu = 7.56 \times 10^7$	$n_2 = 71\%$	Wagner <i>et al.</i> 2009 (ref. 36)
Ag	SiN <sub>x</sub>	$E = 4 \times 10^3$ , $D = 1.5 \times 10^{16}$ , $D = 3.0 \times 10^{16}$ , $D = 4.5 \times 10^{16}$	HRTEM, XS-TEM	$\lambda = 475 \text{ nm}$		Bayle <i>et al.</i> 2015 (ref. 37)
Ag	SiO <sub>2</sub>	$E = 200$ , $D = 2 \times 10^{17}$	Z-Scan	$\lambda = 532 \text{ nm}$	$\chi^{(3)} = 4.0 \times 10^{-8}$	Wang <i>et al.</i> 2008 (ref. 38)
Ag	SiO <sub>2</sub>	$E = 200$ , $D = 2 \times 10^{17}$	Z-Scan	$\lambda = 1064 \text{ nm}$	$\chi^{(3)} = 9.0 \times 10^{-8}$	Cesca <i>et al.</i> 2010 (ref. 39)
Ag	SiO <sub>2</sub>	$E = 130$ , $J = 2$ , $D = 3 \times 10^{16}$ , $T = 800$ , $t = 60$	TEM, Z-scan	$\lambda = 300\text{--}800 \text{ nm}$	$\beta = (-3.42 \pm 0.06) \times 10^{-4}$ , $I_s = (1.02 \pm 0.03) \times 10^7$	
Ag	SiO <sub>2</sub>	$E = 130$ , $J = 2$ , $D = 3 \times 10^{16}$ , $E = 190$ , $J = 0.2$ , $D = 2.5 \times 10^{16}$ (Ar)	TEM, Z-scan	$\lambda = 300\text{--}800 \text{ nm}$	$\beta = (-3.0 \pm 0.4) \times 10^{-4}$ , $I_s = (1.21 \pm 0.05) \times 10^9$	
Ag	72GeS <sub>2</sub> -18Ga <sub>2</sub> S <sub>3</sub> -10CdS	$E = 70$ , $D = 1 \times 10^{16}$	Z-Scan, AFM	$\lambda = 700 \text{ nm}$	$\chi^{(3)} = 1.58 \times 10^{-11}$	Song <i>et al.</i> 2015 (ref. 40)
Ag	72GeS <sub>2</sub> -18Ga <sub>2</sub> S <sub>3</sub> -10CdS	$E = 70$ , $D = 3 \times 10^{16}$	Z-Scan, AFM	$\lambda = 700 \text{ nm}$	$\chi^{(3)} = 4.83 \times 10^{-11}$	
Ag	72GeS <sub>2</sub> -18Ga <sub>2</sub> S <sub>3</sub> -10CdS	$E = 70$ , $D = 3 \times 10^{16}$	Z-Scan, AFM	$\lambda = 700 \text{ nm}$	$\chi^{(3)} = 5.70 \times 10^{-11}$	
Ag	72GeS <sub>2</sub> -18Ga <sub>2</sub> S <sub>3</sub> -10CdS	$E = 70$ , $D = 1 \times 10^{17}$	Z-Scan, AFM	$\lambda = 700 \text{ nm}$	$\chi^{(3)} = 7.58 \times 10^{-11}$	
Ag	72GeS <sub>2</sub> -18Ga <sub>2</sub> S <sub>3</sub> -10CdS	$E = 70$ , $D = 2 \times 10^{17}$	Z-Scan, AFM	$\lambda = 700 \text{ nm}$	$\chi^{(3)} = 2.87 \times 10^{-11}$	
Au	SiO <sub>2</sub>	$E = 190$ , $J = 2$ , $D = 3 \times 10^{16}$ , $T = 800$ , $t = 60$ (air)	TEM, Z-scan	$\lambda = 300\text{--}800 \text{ nm}$	$\beta = (1.7 \pm 0.2) \times 10^{-4}$ , $I_s = (3.2 \pm 0.5) \times 10^8$	Sánchezdena <i>et al.</i> 2013 (ref. 41)
Au	SiO <sub>2</sub>	$E = 190$ , $J = 2$ , $D = 3 \times 10^{16}$ , $J = 0.2$ , $D = 2.5 \times 10^{16}$ (Ar)	TEM, Z-scan	$\lambda = 300\text{--}800 \text{ nm}$	$\beta = (16 \pm 8) \times 10^{-4}$ , $I_s = (1.0 \pm 0.4) \times 10^7$	
Au	$\alpha\text{Al}_2\text{O}_3$	$E = 1.5\text{--}2.0 \times 10^3$ , $D = 2.5\text{--}8 \times 10^{16}$	Z-Scan	$\lambda = 532 \text{ nm}$	$\beta = -4 \times 10^{-12}$ , $I_0 = 3.1 \times 10^{14}$ , $n_2 = 3.1 \times 10^{-15}$	
Au	$\alpha\text{Al}_2\text{O}_3$	$E = 1.5\text{--}2.0 \times 10^3$ , $D = 2.5\text{--}8 \times 10^{16}$	Z-Scan	$\lambda = 355 \text{ nm}$	$\beta = 1.5 \times 10^{-12}$ , $I_0 = 3.36 \times 10^{14}$	
Au	Al <sub>2</sub> O <sub>3</sub>		DFWM, Z-scan	$\lambda = 525 \text{ nm}$ , $\tau = 5.2 \text{ ns}$ , $\nu = 10 \text{ Hz}$ , $I_0 = 5.7 \times 10^6$	$\beta = -1.31 \times 10^{-3}$ , $\chi^{(3)} = 1.58 \times 10^{-7}$ , $[\chi^{(3)}] = 6.25 \times 10^{-8}$ , $\text{Im}[\chi^{(3)}] = -5.33 \times 10^{-7}$ , $\text{Re}[\chi^{(3)}] = 5.03 \times 10^{-7}$ , $n_2 = 7.62 \times 10^{-9}$	Ryaanyanskiy <i>et al.</i> 2007 (ref. 42)
Au	SiO <sub>2</sub>		DFWM, Z-scan	$\lambda = 540 \text{ nm}$ , $\nu = 5.8 \text{ ns}$ , $I_0 = 28 \times 10^6$	$\beta = -0.12 \times 10^{-3}$ , $\chi^{(3)} = 1.38 \times 10^{-7}$ , $[\chi^{(3)}] = 2.7 \times 10^{-8}$ , $\text{Im}[\chi^{(3)}] = -0.067 \times 10^{-7}$ , $\text{Re}[\chi^{(3)}] = 2.97 \times 10^{-9}$	



Table 1 (Contd.)

Metal (ion)	Matrix	Synthesis conditions: energy ( $E$ ), keV, dose ( $D$ ), ion per $\text{cm}^2$ , current density ( $J$ ), $\mu\text{A cm}^{-2}$ , annealing temperature ( $T$ ), $^\circ\text{C}$ , time ( $t$ ) ps	Study	Laser parameters: wavelength ( $\lambda$ ), nm, pulse duration ( $\tau$ ), ps, repetition rate ( $\nu$ ), Hz, intensity ( $I_0$ ), $\text{W cm}^{-2}$ , pulse energy ( $P$ ) mJ	Nonlinear parameters: refract. coeff. ( $n_2$ ), $\text{cm}^2 \text{W}^{-1}$ , absorption coeff. ( $\beta$ ), $\text{cm W}^{-1}$ , satur. intensity ( $I_{\text{sat}}$ ), $\text{W cm}^{-2}$ , $\text{Re}[\chi^{(3)}]$ , $\text{Im}[\chi^{(3)}]$ , $[\chi^{(3)}]$ , esu	Author
Au	ZnO		DFWM, Z-scan	$\lambda = 500 \text{ nm}$ , $\nu = 5.9 \text{ ns}$ , $I_0 = 8 \times 10^6$	$\beta = -0.08 \times 10^{-3}$ , $\chi^{(3)} = 1.47 \times 10^{-7}$ , $[\chi^{(3)}] = 2.46 \times 10^{-8}$ , $\text{Im}[\chi^{(3)}] = -0.044 \times 10^{-7}$ , $\text{Re}[\chi^{(3)}] = -1.47 \times 10^{-7}$ , $n_2 = -1.31 \times 10^{-9}$	Chen <i>et al.</i> 2012 (ref. 43)
Au	BTEAADT (2-thioxo-1,3-dithiole-4,5-dithiolato)		Z-Scan	$\lambda = 532 \text{ nm}$	$\beta = 2.839 \times 10^{-9}$ , $\chi^{(3)} = 9.410 \times 10^{-13}$ , $n_2 = -1.685 \times 10^{-14}$ , $\gamma = 6.770 \times 10^{-31}$ , ( $\gamma$ : hyperpolarizability)	Chen <i>et al.</i> 2012 (ref. 43)
BTEAADT			Z-Scan	$\lambda = 1064 \text{ nm}$	$\chi^{(3)} = 6.638 \times 10^{-13}$ , $n_2 = -1.459 \times 10^{-14}$ , $\gamma = 4.303 \times 10^{-31}$	Torres-Torres <i>et al.</i> 2015 (ref. 44)
Au-NPs	$\text{SiO}_2$	$E = 2 \times 10^3$ , $D = 5 \times 10^{16}$	Z-Scan	$\lambda = 825 \text{ nm}$ , $t = 80 \text{ fs}$	$\beta = -1.51 \times 10^{-7}$ , $[\chi^{(3)}] = 1.21 \times 10^{-10}$ , $n_2 = 8.94 \times 10^{-11}$	Torres-Torres <i>et al.</i> 2015 (ref. 44)
Au-NPs	$\text{SiO}_2$	$E = 2 \times 10^3$ , $D = 5 \times 10^{16}$	Z-Scan	$\lambda = 532 \text{ nm}$ , $t = 1 \text{ ns}$	$[\chi^{(3)}] = 2.2 \times 10^{-9}$	Torres-Torres <i>et al.</i> 2015 (ref. 44)
Si-QDs & Au-NP	$\text{SiO}_2$	$E_1 = 1.5 \times 10^3$ , $D_1 = 2.5 \times 10^{17}$ , $E_2 = 1.5 \times 10^3$ , $D_2 = 8.5 \times 10^{16}$	Z-Scan	$\lambda = 825 \text{ nm}$ , $t = 80 \text{ fs}$	$\beta = -1.26 \times 10^{-6}$ , $[\chi^{(3)}] = 2.6539 \times 10^{-9}$ , $n_2 = 3.37 \times 10^{-11}$	Torres-Torres <i>et al.</i> 2015 (ref. 44)
Si-QDs & Au-NP	$\text{SiO}_2$	$E = 1.5 \times 10^3$ , $D_2 = 8.5 \times 10^{16}$	Z-Scan	$\lambda = 532 \text{ nm}$ , $t = 1 \text{ ns}$	$[\chi^{(3)}] = 3.5 \times 10^{-9}$	Torres-Torres <i>et al.</i> 2015 (ref. 44)
Si-QDs	$\text{SiO}_2$	$E = 1.5 \times 10^3$ , $D = 2.5 \times 10^{17}$ , $T = 1100^\circ\text{C}$ , $t = 90 \text{ min}$ , (50% $\text{H}_2 + 50\% \text{N}_2$ )	Z-Scan	$\lambda = 825 \text{ nm}$ , $t = 80 \text{ fs}$	$\beta = -1.51 \times 10^{-5}$ , $[\chi^{(3)}] = 8.682 \times 10^{-9}$ , $n_2 = 5.55 \times 10^{-11}$	Torres-Torres <i>et al.</i> 2015 (ref. 44)
Si-QDs	$\text{SiO}_2$	$E = 1.5 \times 10^3$ , $D = 2.5 \times 10^{17}$ , $T = 1100^\circ\text{C}$ , $t = 90 \text{ min}$ , (50% $\text{H}_2 + 50\% \text{N}_2$ )	Z-Scan	$\lambda = 532 \text{ nm}$ , $t = 1 \text{ ns}$	$[\chi^{(3)}] = 5.3 \times 10^{-10}$	Torres-Torres <i>et al.</i> 2015 (ref. 44)
Mg	ZnO	$E = 3.24 \times 10^{-3}$	XPS, Z-scan, UV-Vis	$\lambda = 490 \text{ nm}$	$\beta = 58 \times 10^{-6}$ , $\text{Im}[\chi^{(3)}] = 5.91 \times 10^{-11}$ , $\text{Re}[\chi^{(3)}] = 2.78 \times 10^{-11}$ , $n_2 = -2.04 \times 10^{-5}$	Agrawal <i>et al.</i> 2015 (ref. 45)
Mg	$\text{Mg}_{0.09}\text{Zn}_{0.91}\text{Q}$	$E = 3.37 \times 10^{-3}$	XPS, Z-scan, UV-Vis	$\lambda = 550 \text{ nm}$	$\beta = 448 \times 10^{-6}$ , $\text{Im}[\chi^{(3)}] = 41.99 \times 10^{-11}$ , $\text{Re}[\chi^{(3)}] = 3.08 \times 10^{-11}$ , $n_2 = -2.46 \times 10^{-5}$	Torres-Torres <i>et al.</i> 2015 (ref. 45)
Mg	$\text{Mg}_{0.21}\text{Zn}_{0.79}\text{Q}$	$E = 3.49 \times 10^{-3}$	XPS, Z-scan, UV-Vis	$\lambda = 590 \text{ nm}$	$\beta = 455 \times 10^{-6}$ , $\text{Im}[\chi^{(3)}] = 36.41 \times 10^{-11}$ , $\text{Re}[\chi^{(3)}] = 2.50 \times 10^{-11}$ , $n_2 = -2.36 \times 10^{-5}$	Torres-Torres <i>et al.</i> 2015 (ref. 45)
Ni	BTEAADT (2-thioxo-1,3-dithiole-4,5-dithiolato)		Z-Scan	$\lambda = 532 \text{ nm}$	$\chi^{(3)} = 7.84 \times 10^{-13}$ , $n_2 = -1.452 \times 10^{-14}$ , $\gamma = 6.770 \times 10^{-31}$ , ( $\gamma$ : hyperpolarizability)	Chen <i>et al.</i> 2012 (ref. 43)

Nowadays, high-speed optical devices require materials with high values of nonlinear optical parameters such as nonlinear refractive index and nonlinear absorption coefficient.<sup>46</sup> For Cu, with the increasing dose number, the absorption coefficient and the  $\chi^{(3)}$  also increases under the same detection conditions, but the synthesis conditions are different. The results of new investigations on the nonlinear optical parameters of composite materials based on an STO matrix with Cu and Tb nanoparticles are presented. They showed that the effective third-order nonlinear susceptibility is real and has a positive sign. DFWM and Z-scan measurements showed that the value of  $|\chi^{(3)}|$  increased with an increase in the implantation dose. Moreover, the slow component of the relaxation time increased (from 0.1 to 3.58 units for Cu and from 2.52 to 4.15 for Tb nanoparticles).

Ag was implanted into the matrix of  $\text{SiO}_2$ , and the sample was tested at the wavelengths 532 nm and 1064 nm; the sample tested at the lower wavelength had a lower  $\chi^{(3)}$ ; this indicated that this new structure of metal nanoclusters represented a new aspect for the study of the nonlinear optical response. However, if the synthesis conditions were changed, the nonlinear parameters changed together. Furthermore, experiments showed that the nonlinear absorption properties of the samples were affected after Ar irradiation. The results revealed the large nonlinear absorption properties of these systems, as characterized by the concomitance of saturable and reverse saturable absorption.

The third-order nonlinearity of  $72\text{GeS}_2-18\text{Ga}_2\text{S}_3-10\text{CdS}$  chalcogenide glass implanted by Ag ions was studied. The doses for implantation ranged from  $1 \times 10^{16}$  to  $2 \times 10^{17}$  ions per  $\text{cm}^2$ , and the Ag nanoparticles were observed *via* AFM measurements. The third-order nonlinear optical property  $\chi^{(3)}$  was measured by the femtosecond Z-scan technique and showed a maximum value of  $7.58 \times 10^{-11}$  esu.

This indicated that the  $\chi^{(3)}$  enhancement of the implanted samples was due to the formation of Ag nanoparticles. The relation between the implanted dose and the third-order nonlinearity was associated with the enhancement of the local field inside the particles and the interaction between Ag nanoparticles, which could be useful in the fabrication of optical devices by controlling the implanted doses to control the optical nonlinearity in glasses.

## 2.1 Nonlinear optical studies of Au

The third-order NOPS of gold nanoparticles implanted into various matrices ( $\text{Al}_2\text{O}_3$ , ZnO, and  $\text{SiO}_2$ ) have been investigated by the Z-scan method. The nonlinear refractive index, nonlinear absorption coefficient, and the real and imaginary parts of the third-order nonlinear susceptibility have been deduced.<sup>41</sup> According to the results, when the wavelength was between 300 nm and 800 nm, it was clear that the annealing atmosphere could affect the absorption coefficient. The samples annealed in air had a lower absorption coefficient than the sample annealed

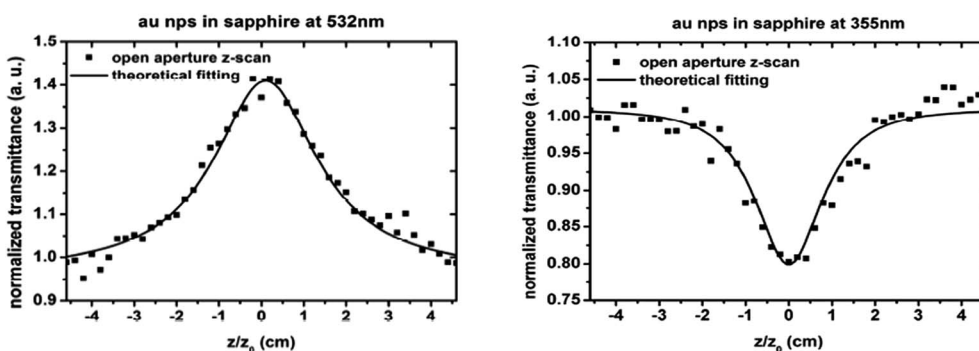


Fig. 3 Nonlinear absorption values for the isotropic Au NPs in  $\alpha\text{-Al}_2\text{O}_3$  at 532 nm and 355 nm. For the first wavelength, the nonlinear optical absorption is negative, whereas it is positive for the second wavelength.<sup>41</sup>

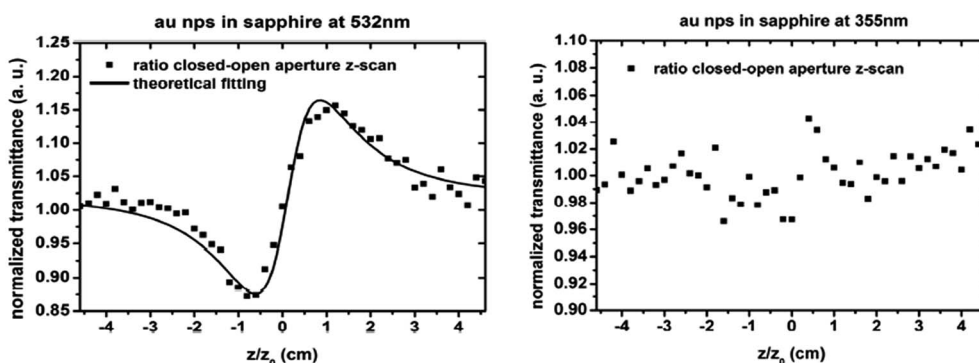


Fig. 4 Nonlinear refraction for the isotropic Au NPs in  $\alpha\text{-Al}_2\text{O}_3$  at 532 nm and 355 nm. For the first wavelength, the nonlinear optical refraction is positive, whereas it is null for the second wavelength.<sup>41</sup>



in Ar. Regarding the nonlinear optical response, it was found that the isotropic sample exhibited negative and positive nonlinear absorption at the wavelengths of 532 nm and 355 nm, respectively (Fig. 3). The nonlinear coefficients were  $\beta = -4 \times 10^{-12} \text{ m W}^{-1}$  with an irradiance value of  $I_0 = 3.1 \times 10^{14} \text{ W m}^{-2}$  for 532 nm, and  $\beta = 1.5 \times 10^{-8} \text{ m W}^{-1}$  with  $I_0 = 3.36 \times 10^{14} \text{ W m}^{-2}$  for 355 nm. Moreover, the nonlinear refraction was found to be positive for 532 nm with a value of  $n_2 = 3.1 \times 10^{-15} \text{ m}^2 \text{ W}^{-1}$ , but it vanished at 355 nm (Fig. 4).

Analysis of the absorption spectra of gold nanoparticles in three matrices showed that the SPR of gold nanoparticles was located in the region of 525 nm for  $\text{Al}_2\text{O}_3$ ,<sup>47</sup> 540 nm for  $\text{ZnO}$ ,<sup>48</sup> and 500 nm for  $\text{SiO}_2$ .<sup>49</sup> The values of the absorption coefficient reported for the matrices under consideration were  $3.3 \times 10^{-16} \text{ cm}^2 \text{ W}^{-1}$  for  $\text{Al}_2\text{O}_3$ ,  $-9 \times 10^{-15} \text{ cm}^2 \text{ W}^{-1}$  for  $\text{ZnO}$ ,<sup>50</sup> and  $2.24 \times 10^{-16} \text{ cm}^2 \text{ W}^{-1}$  for  $\text{SiO}_2$  (ref. 51) at the wavelength of 532 nm. These data were much smaller than those of the composite materials. Thus, it could be concluded that the nonlinear optical response of the composites was derived from the gold nanoparticles.

## 2.2 Nonlinear optical studies of silicon-based materials

Silicon-based optical devices have received significant attention because of their potential application in high-speed signal processing and on-chip communications.<sup>52</sup> The NOPs of silicon have been studied for a long time and have been subsequently characterized with various different techniques; this has resulted in a large range of reported data.<sup>53</sup> The accuracies of these studies were quite essential, which were used to characterize the linear coupling and propagation losses. The nonlinear optical properties of silicon have been utilized in the design of modulators, receivers, and filters in optical communication demonstrations with bit rates in the order of 1 Tbits per s (ref. 54) and in real-time A-to-D converters with the sampling rates of several tens of Gbits per s. A major advantage of the silicon photonic devices is that they can be produced efficiently because of the superiority of the mature silicon processing technology that has been extensively developed to permit low-cost, large-volume electronic circuit production. This makes optical devices compatible with the CMOS technology possible for on-chip integration. The high bandwidths, high speeds, and energy efficiency of optical communication circuits allow on-board communication with performances greatly surpassing those of the electronic alternative.<sup>55</sup>

Si and  $\text{SiO}_2$  multilayers with a dot size of 2.5 nm were prepared, and their microstructures were characterized by cross-sectional transmission microscopy and Raman spectroscopy. A broad photoluminescence band centered at 870 nm was observed. The multi-layer nonlinear optical effect was studied by the Z-scanning technique under the excitation of two laser pulses. Photoluminescence at room temperature was observed at 870 nm; the large Stokes shift between the linear absorption edge and emission band implied that the emission might originate from the recombination of photoexcited carriers *via* interface states with an energy level inside the gap.<sup>56</sup>

In Table 2, the nonlinear absorption coefficient and nonlinear refractive index of the sample upon changing the

**Table 2** The nonlinear optical response of nc-Si/ $\text{SiO}_2$  by changing the excitation pulse duration from ps to fs<sup>56</sup>

Excitation pulse duration	Nonlinear absorption coefficient ( $\beta$ )/(cm $\text{W}^{-1}$ )	Nonlinear refractive index ( $n_2$ )/(cm <sup>2</sup> $\text{W}^{-1}$ )
Picosecond laser (ps)	$-3.1 \times 10^{-6}$	$-1.3 \times 10^{-10}$
Femtosecond laser (fs)	$1.1 \times 10^{-7}$	$-1.5 \times 10^{-12}$

excitation pulse time from ps to fs are shown. When the sample was excited by a picosecond laser, it showed saturation absorption with  $\beta = -3.1 \times 10^{-6} \text{ cm W}^{-1}$  and  $n_2 = -1.3 \times 10^{-10} \text{ cm}^2 \text{ W}^{-1}$ . The negative nonlinear refraction could be attributed to the free carrier dispersion effect. The saturation absorption can be attributed to the single photon transition process between the valence band and the interface state. However, when the sample was excited by a femtosecond laser, the sample showed reverse saturation absorption with  $\beta = 1.1 \times 10^{-7} \text{ cm W}^{-1}$  and  $n_2 = -1.5 \times 10^{-12} \text{ cm}^2 \text{ W}^{-1}$ . The difference in the free carrier density generated by two pulses laser played a great role in the diverse nonlinear response.

A tunable nonlinear absorption and nonlinear refraction response in Si/ $\text{SiO}_2$  multilayers have been observed during the transition process from the amorphous to nanocrystalline phases under femtosecond excitation at 800 nm by Zhang *et al.*<sup>57</sup> They suggested that the two-photon absorption process dominates in the samples mainly containing amorphous Si phases, whereas the phonon-assisted one-photon transition process between the valence band and interface states dominates the nonlinear optical properties in nc-Si/ $\text{SiO}_2$  multilayers. This suggests that nc-Si/ $\text{SiO}_2$  multilayers can be applied in highly sensitive photonic devices such as optical switches and Q-switch lasers.

Silicon nanocrystals are widely used in photonics, with their most important characteristics being two aspects: (1) they can be used as a nonlinear material in various devices, *e.g.*, in bistable optical cavities, in waveguide optical mode monitors based on two-photon excited luminescence detection, and in wavelength shifters using four-wave mixing; (2) they can be used as an entropy source for quantum random number generation, which is the key device for cryptography. Quantum size effects change the properties of Si-nc with respect to bulk silicon; this allows enlargement of the phenomenology and the application spectrum of silicon photonics.<sup>58</sup>

## 2.3 Nonlinear optical studies of Ag

Due to their large third-order nonlinear optical properties caused by surface plasmon resonance and the quantum size effect, noble metallic nanostructures embedded on a transparent dielectric matrix have been attracting significant attention<sup>59,60</sup> as promising materials for all-optical signal processing devices.<sup>61,62</sup> Consequently, a few technologies have been developed such as laser ablation,<sup>63,64</sup> chemical synthesis, ion implantation,<sup>65-67</sup> and sol-gel methods. The quantum size effect



will make the material show a more excellent performance, different from that of other noble metallic NPs, and silver NPs experience a lower intrinsic loss of plasmonic energy at visible frequencies that gives rise to SPR.<sup>68</sup> Moreover, in further research, it has been indicated that the shape of the metallic NPs and their dielectric environment determine the surface frequency and consequently the nonlinear optical properties.<sup>65,69,70</sup>

NF-RGO (non-covalent functionalized reduced graphene oxide) and NF-RGO/Ag-NPs (NF-RGO decorated with various concentration of silver nanoparticles) were prepared by Sakho *et al.*<sup>71</sup> The open aperture Z-scan technique was used to investigate the nonlinear optics. Tables 3 and 4 show the numerical calculated values of the nonlinear optical parameters at  $\lambda = 800$  nm with femtosecond and  $\lambda = 532$  nm with nanosecond excitation regimes, respectively. The lower optical limiting threshold was attributed to the charge or energy transfer between NF-RGO and Ag-NPs in the NF-RGO/Ag-NP hybrid. In nonlinear optical absorption, the lower the optical threshold value, the better the optical limiting properties.

In the fs regime,  $\beta$  was found to increase with the decreasing concentration of AgNPs. The higher the  $\beta$  value, the better the OL properties of the corresponding material. In addition, in the fs regime, the samples exhibited the same optical limiting behavior as in the ns regime. The optical limiting thresholds were found to be lower in the fs excitation regime as compared

to those in the ns excitation regime. This was attributed to the prominent plasmon band bleaching of AgNPs when excited at 532 nm. Furthermore, the nonlinear optical analyses showed that the optical properties of the NF-RGO/Ag-NP hybrid were size dependent. This finding provided a good understanding of the graphene-based nonlinear materials and also indicated that the as-synthesized materials could be potential candidates for optical limiting applications. Moreover, Li *et al.*<sup>68</sup> have observed that the nonlinear absorption coefficients significantly depend on the excitation energy and can be modulated by varying the fluence of  $\text{Ag}^+$  ions.

### 3 NOPs of components synthesized by different methods

#### 3.1 Vacuum electron-beam co-evaporation method

In the past few decades, the nanostructures of direct band gap semiconductors, such as Si and Ge, have been extensively studied because of their potential applications in nanophotonics, with their main advantage being their compatibility with conventional integrated circuit technology. Amorphous and crystallized Ge nanostructures embedded in an  $\text{Al}_2\text{O}_3$  dielectric were synthesized by a vacuum electron-beam co-evaporation method.<sup>72</sup> A clear blue-shift of the absorption edge and a large third-order nonlinear optical susceptibility due to the quantum confinement effect were observed.

Herein, three samples with different Ge concentrations ((A)  $\text{Ge}/\text{Al}_2\text{O}_3 = 1/5$ , (B)  $\text{Ge}/\text{Al}_2\text{O}_3 = 2/5$ , and (C)  $\text{Ge}/\text{Al}_2\text{O}_3 = 5/5$ ) were synthesized, and the thickness was about 200 nm. The band gap of the semiconductor was increased due to the quantum confinement effect, which caused a blue-shift of the absorption band edge and optical luminescence.<sup>73,74</sup> Moreover, this effect caused an enhancement of the third-order optical nonlinearity. Optical absorption measurements were performed in the visible to near-infrared range at room temperature. Fig. 5 shows the optical absorption spectra of Ge +  $\text{Al}_2\text{O}_3$  films on quartz substrates. It was clear that all these samples exhibited a blue-shift of the band gap because the band gap of the bulk amorphous Ge ( $\alpha\text{-Ge}$ ) was about 0.9 eV. According to the indirect transition equation,

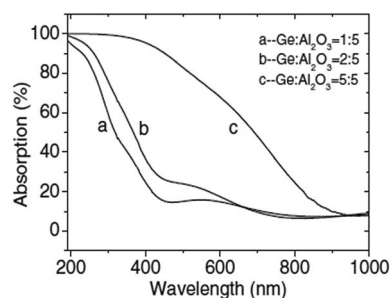
$$\alpha(h\nu) = \alpha(h\nu - E_g)^2/h\nu \quad (11)$$

**Table 3** Nonlinear optical parameters of the samples under 800 nm fs excitation<sup>71</sup>

Samples	$\beta$ ( $10^{-15} \text{ m W}^{-1}$ )	$I_s$ ( $10^{16} \text{ W m}^{-2}$ )	Optical limiting threshold ( $\text{J cm}^{-2}$ )
NF-RGO	2.5	3.5	1.6
NF-RGO/ Ag-NPs (0.1 M)	6.7	2.5	1.7
NF-RGO/ AgNPs (0.3 M)	6.0	2.5	1.7
NF-RGO/ AgNPs (0.5 M)	4.8	3.9	1.7
NF-RGO/ AgNPs (1 M)	5.8	3.5	1.6

**Table 4** Nonlinear optical parameters of the samples under 532 nm ns excitation<sup>71</sup>

Samples	$\beta$ ( $10^{-10} \text{ m W}^{-1}$ )	$I_s$ ( $10^{11} \text{ W m}^{-2}$ )	Optical limiting threshold ( $\text{J cm}^{-2}$ )
NF-RGO	2.8	3.8	8.3
NF-RGO/ AgNPs (0.1 M)	7.99	4	4.3
NF-RGO/ AgNPs (0.3 M)	7.99	3.8	5.9
NF-RGO/ AgNPs (0.5 M)	7.99	3.7	7.2
NF-RGO/ AgNPs (1 M)	13.9	2.8	8.1



**Fig. 5** Optical absorption spectra of Ge nanocrystals embedded in  $\text{Al}_2\text{O}_3$  films on quartz substrates (synthesized at room temperature; thickness: 200 nm).<sup>72</sup>







Fig. 6 Z-Scan plots of the Ge nanocrystals embedded in an  $\text{Al}_2\text{O}_3$  film ( $\text{Ge} : \text{Al}_2\text{O}_3 = 1 : 5$ ): (a) closed aperture and (b) open aperture. The solid curves are the theoretical fits.<sup>72</sup>

$E_g$  was calculated to be about 2.4, 2.2, and 1.2 eV for the samples A, B, and C, respectively.

The third-order nonlinear optical characteristics of sample A were determined by the Z-scan method at a wavelength of 532 nm and a pulse width of 10 ns. The closed aperture plot (Fig. 6(a)) shows a distinct valley-peak configuration typical of the positive nonlinear effects (self-focusing). The nonlinear optical refractive index  $n_2$  defined as

$$n = n_0 + n_2 I \quad (12)$$

was calculated using the following equation:

$$n_2 \text{ (m}^2 \text{ W}^{-1}\text{)} = 1.232 \lambda \tau \omega_0^2 \Delta T_{\text{P-V}} / (1 - s)^{0.25} L_{\text{eff}} E \quad (13)$$

where  $n_0$  is the linear refractive index of the samples,  $\lambda$  is the wavelength of light,  $\tau$  is the laser pulse width,  $L_{\text{eff}}$  is the effective thickness,  $E$  is the pulse energy at the focal spot, and  $\Delta T_{\text{P-V}} = 0.72$ . Fig. 6(b) shows the normalized open aperture Z-scan transmission for sample A. A symmetric inverted bell-shaped curve can be observed with a minimum at the focus, which indicates a two-photon or multiphoton absorption process.

### 3.2 Melt-quenching technique

Glasses containing metal nanoparticles exhibit an ultrafast electron response within a few of picoseconds, which also have large third-order optical nonlinearities for a large local-field enhancement factor and strong resorption of the surface plasmon.<sup>75–77</sup> Thus, these materials have been intensively studied in the last few decades as they are widely used in ultrafast all-optical switching, ultrafast imaging, optical telecommunications, signal processing, and so on.<sup>78,79</sup> Glasses containing metal nanoparticles are generally fabricated by melt quenching,

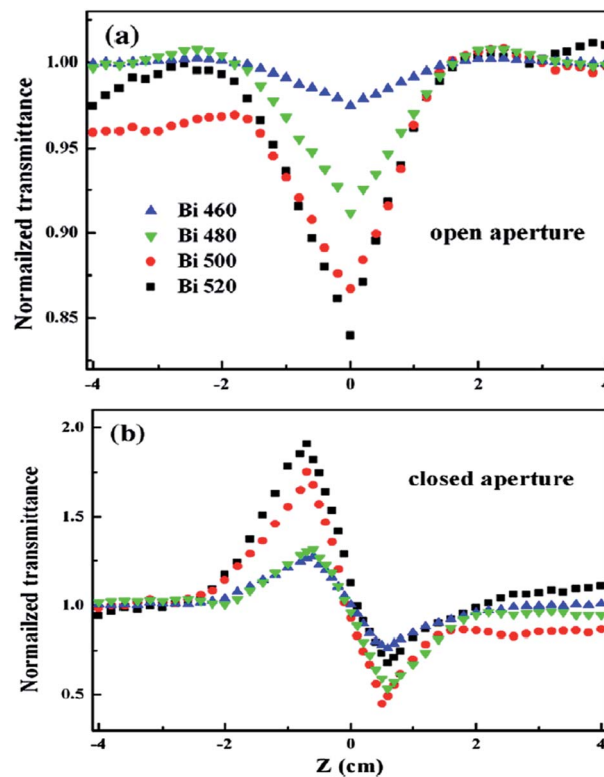


Fig. 7 Typical Z-scan normalized transmittance curves for (a) open and (b) closed aperture measurements.<sup>85</sup>

the sol-gel method, sputtering, ion exchange, ion implantation, or femtosecond laser irradiation.<sup>80–82</sup> It is typically really difficult to control Bi nanoparticles formation in glasses. Thus, Lin *et al.*<sup>83</sup> proposed a simple method, whereby they used Al to reduce Bi ions to atomic state Bi, and the atomic or molecular Bi gathered together. Bi grows and forms nanocrystals during the subsequent heat treatment. The Z-scan technique was used to investigate the nonlinear optical properties of the Bi nanoparticle composite glasses.

Typical Z-scan normalized transmittance curves for open and closed aperture measurements of these Bi glasses are shown in Fig. 7. The nonlinear index  $\gamma$ , nonlinear refractive index  $n_2$ , nonlinear absorption coefficient  $\beta$ , and third-order optical susceptibility  $\chi^{(3)}$  of all the samples were obtained by applying nonlinear curve fitting, and the obtained third-order optical nonlinearities are shown in Table 5. It can be found that the third-order optical susceptibility  $\chi^{(3)}$  increases as the heat treatment temperature increases.

Table 5 Third-order nonlinearities of the Bi-doped glasses<sup>85</sup>

Sample	$n_2$ ( $10^{-9}$ esu)	$\beta$ (cm $\text{GW}^{-1}$ )	$\chi^{(3)}$ ( $10^{-7}$ esu)
Bi 460	0.68	0.82	0.39
Bi 480	1.78	3.99	1.01
Bi 500	3.42	6.66	1.95
Bi 520	4.37	9.54	2.49



**Table 6** Dependence of  $\alpha_0$ ,  $\beta$ ,  $n_2$ ,  $T$ , and  $W$  for Au PPAs with different sizes at the intensity of  $55 \text{ GW cm}^{-2}$ .<sup>89</sup>

Size (nm)	$\alpha_0$ ( $\text{cm}^{-1}$ )	$\beta$ ( $\text{cm GW}^{-1}$ )	$n_2$ ( $\text{cm}^2 \text{GW}^{-1}$ )	$W$	$T$
37	$1.4 \times 10^3$	54	-0.035	7.5	0.12
70	$1.5 \times 10^3$	29	-0.032	5.6	0.07
140	$1.5 \times 10^3$	5	0.17	4.5	0.02
190	$1.7 \times 10^3$	0	0.02	4.1	0

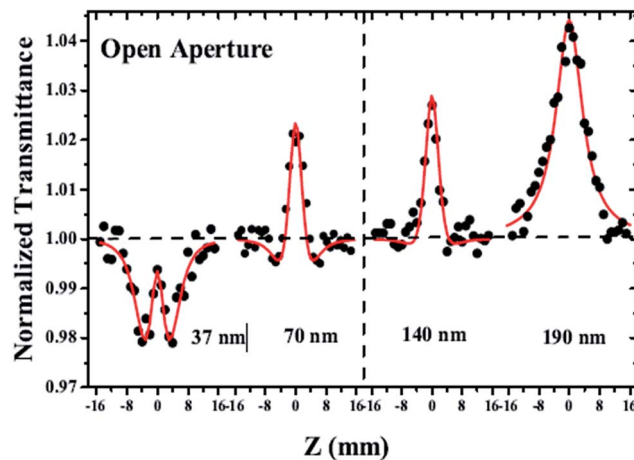
### 3.3 Nanosphere lithography

A systematic study of the size-related nonlinear optical properties of triangular Au particles was reported, where triangular Au nanoparticle arrays with four sizes (37 nm, 70 nm, 140 nm, and 190 nm) were fabricated on  $\text{SiO}_2$  substrates using nanosphere lithography. There are many ways, such as chemical methods<sup>84–86</sup> and sputtering method,<sup>87</sup> to fabricate particles of different sizes. Nanosphere lithography has been developed from natural lithography by Hulteen *et al.*<sup>88</sup> in 1995 and proven to be a powerful tool to fabricate periodic particle arrays (PPAs) with a tunable shape and size, which has made it possible to quantitatively study the optical properties of nanoparticles (Table 6).

Fig. 8 shows the absorption spectra of Au PPAs of different sizes. It can be seen that the absorption peaks due to the SPR are located at 552 nm, 566 nm, 580 nm, and 606 nm for the 37 nm, 70 nm, 140 nm, and 190 nm sizes, respectively. With the increasing particle size, the absorption peak shifts to longer wavelength, and when  $\lambda = 800 \text{ nm}$ , the linear absorption coefficient significantly increases. These size-related linear absorption properties lead to competition between two nonlinear mechanisms: the interband transition through the dominant two-photon absorption (TPA) process and the intra-band contribution ascribed to the saturated absorption (SA) process with excitation at the edge of the SPR band for  $\lambda = 800 \text{ nm}$ .<sup>89</sup>



**Fig. 8** Absorption spectra of Au periodic particle arrays with SPR peaks at 552 nm, 566 nm, 580 nm, and 606 nm. As the sizes increase, the SPR peaks shift to the longer wavelength.<sup>89</sup>



**Fig. 9** The OA Z-scan results of 4 samples at the exciting intensities at the focal point  $I_0 = 55 \text{ GW cm}^{-2}$ . The solid lines show the theoretical best fit.<sup>89</sup>

Fig. 9 shows the typical OA results of the four samples. The black dots indicate the experimental data and the solid lines present the theoretical fits. The laser intensities were calculated to be  $55 \text{ GW cm}^{-2}$  by  $I_0 = E_0/\pi\omega^2\tau$ . At the repetition rate of 1 kHz, the accumulative thermal effects can be neglected. The curve of the 37 nm-sized sample showed a TPA process with an additional SA component. For the curves of the samples sized 70 nm and 140 nm, the TPA component turned weaker and SA became dominant. The curve of the 190 nm sized sample showed only the SA component. The differences are because the absorption in the excitation region is much weaker than that at 400 nm for the Au PPA sized 37 nm, and herein, the interband to the TPA process plays a key role. However, the absorption at 800 nm is larger than that at 400 nm for the Au PPA sized 190 nm; this is because the SA process becomes dominant. The curves of the samples sized 70 nm and 140 nm showed a transition in this variation of the two nonlinear mechanism contributions.

Fig. 10 shows the CA Z-scan data for the Au PPAs at  $I_0 = 55 \text{ GW cm}^{-2}$ . For the Au PPAs sized 37 nm and 70 nm, a self-defocusing effect occurs, where the sign of nonlinear refraction is negative. For the Au PPAs sized 140 nm and 190 nm, self-defocusing occurs, with a positive nonlinear refraction. For the Au PPAs sized 37 nm and 70 nm, the nonlinear refraction index decreases due to the dominant interband transition caused by the TPA process. As the particle size increases, the SA process becomes dominant. The refractive index increases due to the excited electrons; this results in self-focusing. It is worth nothing that there are two clear deviations from the theoretical values of data for the 70 nm sized sample; this indicates a transition process from self-defocusing to self-focusing (Table 7).

### 3.4 Organometallic pyrolysis method

In recent years, the NOPs of nanoparticles have attracted significant interest due to their various applications in



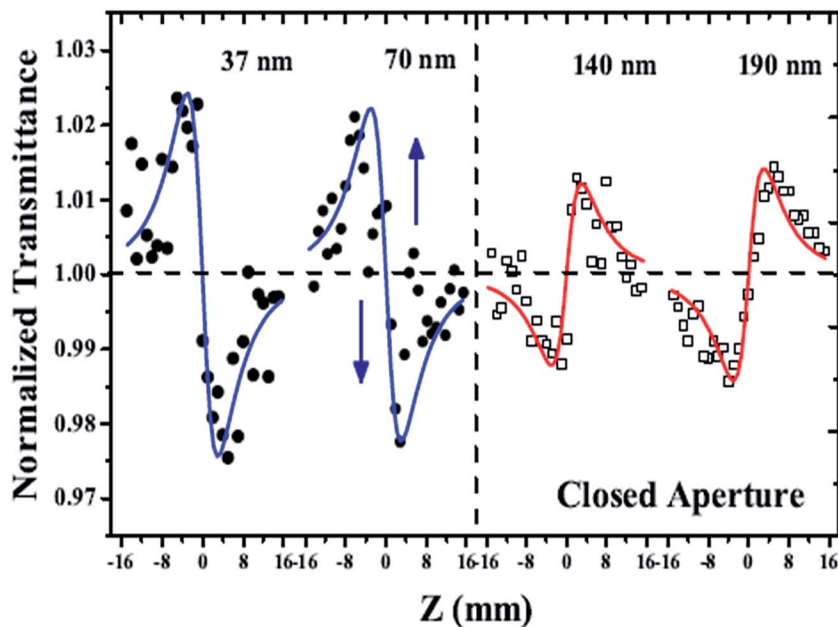


Fig. 10 The CA Z-scan results of the 4 samples at exciting intensities at the focal point  $I_0 = 55 \text{ GW cm}^{-2}$ . The solid lines show the theoretical best fit.<sup>89</sup>

Table 7 Nonlinear absorption parameters of CdSe: Au of different QD sizes<sup>101</sup>

CdSe size (nm)	$\alpha_0$ ( $\text{cm}^{-1}$ )	$L_{\text{eff}}$ (cm)	$\beta$ ( $10^{-10} \text{ m W}^{-1}$ )	$\text{Im } \chi^{(3)}$ ( $10^{-12} \text{ esu}$ )
3.37	3.77	0.18	9.8	24
3.54	4.07	0.17	8.5	20
3.91	4.72	0.16	5.0	12
4.56	4.49	0.16	1.8	4.3
5.07	3.34	0.19	1.4	3.4

optoelectronic and biological fields such as in optical limiting,<sup>90,91</sup> multiphoton imaging,<sup>92,93</sup> all-optical switching,<sup>94,95</sup> cell biology, and cancer-therapy.<sup>96-98</sup> The combination of

nanometals and semiconductors has acquired particular interest for basic sciences and technological applications.<sup>99,100</sup>

In the study reported by Salah *et al.*,<sup>101</sup> Au NPs and nano-composite of Au–CdSe of different NPs sizes and concentrations were prepared.

The effect of the nanosize of CdSe on the nonlinear absorption was investigated, as shown in Fig. 11. At the excitation intensities  $I_0 = 118 \mu\text{J}$ , the transmission *via* NPs of different sizes increases near the focus beam ( $z = 0$ ). This is an indication that the absorption of the ground state is higher than that of the excited states. The SA decreases as the QDs size increases. This decrease can be clearly seen in Table 8, where the nonlinear absorption coefficients are summarized together with different parameters used in the fitting process. This decrease is attributed to the decrease in absorption associated with the increase in the QD size.

The effect of the concentrations of NPs on the nonlinear absorption behavior of NPs was examined, as shown in Fig. 12. OA Z-scan transmissions at a laser energy of  $195 \mu\text{J}$  ( $0.31 \text{ GW cm}^{-2}$ ) were carried out for different concentrations of Au–CdSe at CdSe QDs sized 3.54 nm. The observed SA nonlinear behavior



Fig. 11 OA Z-scan of Au–CdSe of different sizes and excitation intensities  $I_0 = 118 \mu\text{J}$ .<sup>101</sup>

Table 8 Nonlinear absorption parameters for CdSe: Au at different concentrations<sup>101</sup>

Conc. ratio of QDs sized 3.54 nm	$\alpha_0$ ( $\text{cm}^{-1}$ )	$L_{\text{eff}}$ (cm)	$\beta$ ( $10^{-10} \text{ m W}^{-1}$ )	$\text{Im } \chi^{(3)}$ ( $10^{-12} \text{ esu}$ )
1/2.5	12.43	0.079	29	69.9
1/5	6.72	0.13	12.7	30.5
1/10	4.04	0.17	4.96	11.9
1/20	2.16	0.22	2.19	5.26
1/40	1.16	0.25	0.908	2.19
1/80	0.61	0.27	0.311	0.075



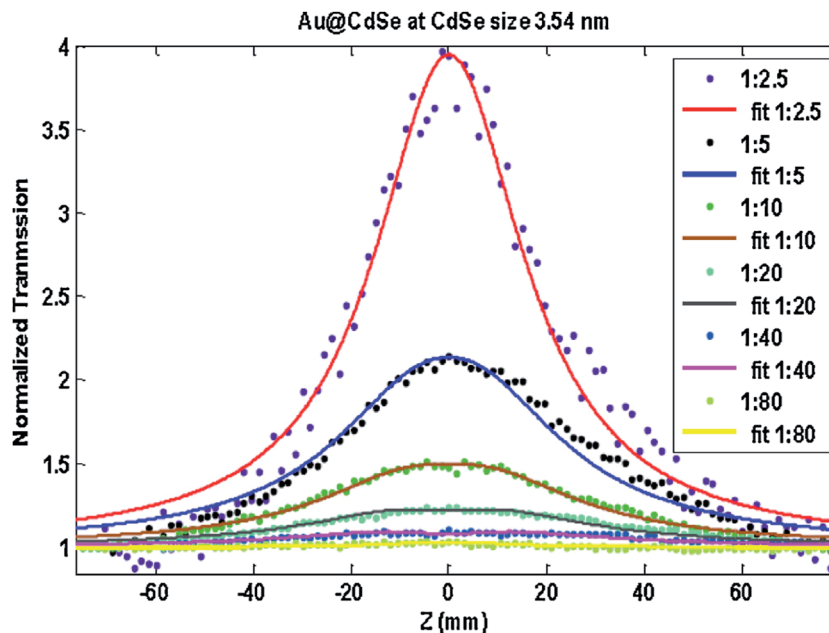


Fig. 12 OA Z-scan transmission of Au–CdSe at  $0.31 \text{ GW cm}^{-2}$  for different concentrations of CdSe (3.54 nm).<sup>101</sup>

Table 9 Values of the linear absorption coefficients ( $\alpha$ ) and nonlinear parameters extracted from the Z-scan measurements<sup>108</sup>

Sample	$\alpha$ ( $\text{cm}^{-1}$ )	$\beta$ ( $\times 10^{-6}$ ) ( $\text{m W}^{-1}$ )	$n_2$ ( $\times 10^{-9}$ ) ( $\text{m}^2 \text{ W}^{-1}$ )	$\chi^{(3)}$ ( $\times 10^{-10}$ ) ( $\text{m}^2 \text{ V}^{-2}$ )	$dn/dT$ ( $\times 10^{-8} \text{ K}^{-1}$ )
A	449	1.51	3.02	1.88	3.8
B	553	1.77	5.31	4.48	5.4
C	629	1.85	8.87	9.45	7.9
D	706	1.74	9.08	11.23	7.3

was found to increase as the concentration of NPs increased. The variation in the calculated nonlinear absorption coefficients of Au–CdSe at different concentrations is summarized in Table 8.

### 3.5 Colloidal chemical synthesis

Among the II–VI semiconductors, CdSe is one of the promising semiconducting materials that has been studied for application in solar cells,<sup>102</sup> thin film transistors,<sup>103</sup> gamma-ray detectors,<sup>104</sup> memory devices,<sup>105</sup> and optoelectronic<sup>106</sup> and biological applications.<sup>107</sup> Polymer nanocomposites (PNCs) are a class of relatively new materials with ample potential applications. Herein, the pure CdSe PNC was marked as sample A. Ag–CdSe hybrid PNC was prepared *via* a colloidal chemical synthesis technique by mixing the required amounts of Ag colloids with CdSe PNC.<sup>108</sup> The volume fraction of Ag was changed by keeping the volume of CdSe PNC constant. The prepared nanocomposite samples had the composition Ag( $x$ )–CdSe PNC, with  $x = 0.1, 0.2,$  and  $0.3\%$ , and these samples were marked as B, C, and D, respectively. The nonlinear optical properties of the nanocomposite thin films were investigated using the Z-scan technique with a 2 mW CW He–Ne laser operating at 632.8 nm.

Thermo-optical coefficient ( $dn/dT$ ) was calculated using the following relation:<sup>109</sup>

$$\frac{dn}{dT} = -\frac{\lambda\kappa}{P\alpha L_{\text{eff}}} \Delta\phi_0 \quad (14)$$

where  $\kappa$  is the thermal conductivity ( $0.09 \text{ W cm}^{-1} \text{ K}^{-1}$ ) and  $\Delta\phi_0$  is the on-axis phase shift. The calculated values of  $dn/dT$  are provided in Table 9. The refractive index of the nanocomposites increases with the increasing Ag content. In addition, with the increasing Ag content, the value of  $dn/dT$  increases; this indicates the increased amount of energy absorbed by PNC and hence its nonlinear parameters. The increase in the value of  $\chi^{(3)}$  for the Ag–CdSe hybrid PNC film as compared to that of the pure CdSe PNC film may be due to the SPR resulting from the doped silver metal, which largely enhances the local field near the particle's surface.<sup>110</sup> Wang *et al.*<sup>111</sup> also observed a dramatic increase (eight times) in the  $n_2$  value for Ag/LiNbO<sub>3</sub> as compared to that in pure LiNbO<sub>3</sub> due to the surface plasmon interactions.

## 4 Conclusions

Nonlinear optics was established soon after the invention of the laser in 1960. It has a long development history of about half a century, and its research field has been continuously developing in depth and breadth. Nonlinear optics studies involve various kinds of nonlinear effects in the interactions between a laser and matter; the research contents include two aspects: one is the study of how to change the macroscopic parameters, such as susceptibility, dielectric coefficient, refractive index, and absorption coefficient, of the matter in the laser with an aim to realize the control of matter by light. The other aspect involves study of how pump-laser-induced matter changes can affect the parameters, such as frequency, power, wave vector,



pulse, pulse width, and propagation direction, of the signal light propagating in the matter, with an aim to realize control of light with light. To date, nonlinear optics has gained extensive applications. For example, it has been widely used in laser technology to change the frequency of the laser and compress the pulse width of the laser; it has also been used to fabricate various optoelectronic devices such as lasers, optical amplifiers, and optical storage and optical switches; it has made an important contribution to the development of nonlinear optical communications and space optical communication; it has been using the nonlinear optical limiting technology to fight against the blinding laser weapons; the nonlinear optics is a useful tool for new materials; and it is also a necessary tool for measuring the nonlinear optical parameters of materials. Devices based on nonlinear optics are compatible with quantum memories and fiber optic communications, as well as with silicon integrated circuit semiconductor technology. These new developments can play a key role in realizing compact, low-cost, and practical sources of complex quantum optical states on a chip, which will ultimately enable quantum technologies to have a significant impact on our society.

There are several different methods and techniques for determining the nonlinear optical response of a material. Z-Scan is one of the simpler experimental methods to be employed and interpreted. Although there are plenty of available methods, it is rare that any single experiment will completely determine the physical processes behind the nonlinear response of a given material. Z-Scan has only recently been introduced. The use of the Z-scan technique as both an absolutely calibrated method for determining standards and as a relative measurement method is increasing. The Z-scan signal can provide useful information on the order of the nonlinearity as well as its sign and magnitude.

Ion implantation has been proven in recent studies as a reliable technique to tune the properties of bulk materials, thin films, nanostructure materials, and biocompatible materials for specific applications. However, the material properties can be altered depending on the proper selection of ion species, ion energy, substrate temperature, and ion fluencies. Compared to other techniques for modifying the surface of the material, ion implantation can obtain the specified characteristics that are needed; therefore, it has wider application prospects. Organic and inorganic composite nonlinear optical materials will likely become a new development trend for nonlinear optical materials due to the powerful theory and technical feasibility behind them. With continuous study, if some problems can be better solved, such as external electric field polarization and the organic and inorganic composite temperature contradiction, a number of high efficiency, high strength, and fast response materials will be developed. This will also support further development of nonlinear optics with regard to device development and theoretical research.

## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 R. W. Boyd and B. R. Masters, Nonlinear optics, *J. Biomed. Opt.*, 2009, **14**(2), 029902.
- 2 S. R. Marder, J. E. Sohn and G. D. Stucky. *Materials for Nonlinear Optics Chemical Perspectives.*; ACS Symp. Ser., American Chemical Society, 1991.
- 3 S. S. Zahraa, S. A. Raad and J. T. Khawla, Study of the Nonlinear Optics Properties of Lithium Triborate Crystal by Using Z-Scan Technique, *Int. J. Sci. Res.*, 2016, 1683–1687.
- 4 P. A. Franken, A. E. Hill, C. W. Peters, *et al.*, Generation of Optical Harmonics, *Phys. Rev. Lett.*, 1961, **7**(4), 118–119.
- 5 T. H. Maiman, Stimulated Optical Radiation in Ruby, *Nature*, 1960, **187**(4736), 493–494.
- 6 C. G. Schroer and B. Lengeler, *Handbook of Lasers and Optics*, 2012, pp. 1461–1474.
- 7 R. L. Byer and A. Piskarskas, Optical Parametric Oscillation and Amplification Introduction, *J. Opt. Soc. Am. B*, 1993, **10**(3), 2148.
- 8 G. A. Askar'Yan, The self-focusing effect, *Phys.-Usp.*, 1973, **16**(5), 680.
- 9 D. W. Forslund, J. M. Kindel and E. L. Lindman, Theory of stimulated scattering processes in laser-irradiated plasmas, *Phys. Fluids*, 1975, **18**(8), 1002–1016.
- 10 U. Kreibig and M. Vollmer. *Optical Properties of Metal Clusters[M]/Optical properties of metal clusters/*, Springer-Verlag, 1995, pp. 278–279.
- 11 B. Raneesh, I. Rejeena, P. U. Rehana, *et al.*, Nonlinear optical absorption studies of sol-gel derived Yttrium Iron Garnet ( $Y_3Fe_5O_{12}$ ) nanoparticles by Z-scan technique, *Ceram. Int.*, 2012, **38**(3), 1823–1826.
- 12 J. U. Seongmin, P. R. Wateker, S. Jeong, *et al.*, Nonlinear optical properties of Zinc doped germane-silicate glass optical fiber, *J. Nonlinear Opt. Phys. Mater.*, 2011, **19**(04), 791–799.
- 13 Y. Wang, C. Y. Lin, A. Nikolaenko, *et al.*, Four-wave mixing microscopy of nanostructures, *Adv. Opt. Photonics*, 2010, **3**(1), 1–52.
- 14 P. Ren, H. Fan and X. Wang, Electrospun nanofibers of ZnO/BaTiO<sub>3</sub>, heterostructures with enhanced photocatalytic activity, *Catal. Commun.*, 2012, **25**(8), 32–35.
- 15 S. Krishnan, C. S. S. Sandeep, R. Philip, *et al.*, Two-photon assisted excited state absorption in multiferroic YCrO<sub>3</sub> nanoparticles, *Chem. Phys. Lett.*, 2012, **529**(5), 59–63.
- 16 S. Hu, C. Melton and D. Mukherjee, A facile route for the synthesis of nanostructured oxides and hydroxides of cobalt using laser ablation synthesis in solution (LASIS), *Phys. Chem. Chem. Phys.*, 2014, **16**(43), 24034–24044.



- 17 S. Basak and R. Chandrasekar, Passive optical waveguiding organic rectangular tubes: tube cutting, controlling light propagation distance and multiple optical out-puts, *J. Mater. Chem. C*, 2013, **2**(8), 1404–1408.
- 18 A. V. Emelyanov, M. V. Khenkin, A. G. Kazanskii, *et al.*, Femtosecond laser induced crystallization of hydrogenated amorphous silicon for photovoltaic applications, *Thin Solid Films*, 2014, **556**(4), 410–413.
- 19 R. Udayabhaskar, M. S. Ollakkan and B. Karthikeyan, Preparation, optical and non-linear optical power limiting properties of Cu, CuNi nanowires, *Appl. Phys. Lett.*, 2014, **104**(1), 958.
- 20 T. Jiang, G. Qin, W. Qin, *et al.*, Passively Q-switched erbium-doped fiber laser based on gold nanorods, *Optik*, 2014, **125**(19), 5789–5793.
- 21 C. Tang, Q. Zheng, H. Zhu, *et al.*, Two-photon absorption and optical power limiting properties of ladder-type tetraphenylene cored chromophores with different terminal groups, *J. Mater. Chem. C*, 2013, **1**(9), 1771–1780.
- 22 H. Fan, X. Wang, Q. Ren, *et al.*, Investigation of the nonlinear absorption and optical limiting properties of two [Q][Cu(CS)] compounds, *Opt. Laser Technol.*, 2010, **42**(5), 732–736.
- 23 L. W. Tutt and T. F. Boggess, A review of optical limiting mechanisms and devices using organics, fullerenes, semiconductors and other materials, *Prog. Quantum Electron.*, 1993, **17**(4), 299–338.
- 24 K. N. Chopra, A short note on technical analysis of excimer lasers, their optimization for laser corneal refractive surgery, and novel applications, *Latin-American Journal of Physics Education*, 2014, **8**.
- 25 N. Pisitpipathsin, P. Kantha, K. Pengpat, *et al.*, Effect of  $\text{KNbO}_3$ , on physical and electrical properties of lead-free  $\text{BaTiO}_3$ , ceramic, *Ceram. Int.*, 2015, **41**(3), 3639–3646.
- 26 A. L. Stepanov, Laser annealing of composite materials with metal nanoparticles[M]/high-power and femtosecond lasers, in *High-Power and Femtosecond Lasers*, ISBN 978-1-60741-009-6, 2009, pp. 27–70.
- 27 L. Nicolais and G. Carotenuto, Metal-Polymer Nanocomposites, *Mater. Today*, 2005, **8**(2), 174–182.
- 28 F. Gonella and P. Mazzoldi, *Handbook of Nanostructured Materials & Nanotechnology*, 2000, vol. 4, ch. 2–Metal nanocluster composite glasses, pp. 81–158.
- 29 A. L. Stepanov, Ion-synthesis of silver nanoparticles and their optical properties, *Langmuir*, 2010, **26**(6), 4176–4183.
- 30 U. Kreibig and M. Vollmer. *Optical Properties of Metal Clusters[M]/Optical Properties of Metal Clusters/*, Springer-Verlag, 1995, pp. 278–279.
- 31 D. Ricard, P. Roussignol and C. Flytzanis, Surface-mediated enhancement of optical phase conjugation in metal colloids., *Opt. Lett.*, 1985, **10**(10), 511–513.
- 32 M. Sheik-Bahae, A. A. Said and E. W. Van Stryland, High-sensitivity, single-beam  $n(2)$  measurements, *Opt. Lett.*, 1989, **14**(17), 955.
- 33 M. Sheik-Bahae, D. J. Hagan and E. W. Van Stryland, Dispersion and band-gap scaling of the electronic Kerr effect in solids associated with two-photon absorption, *Phys. Rev. Lett.*, 1990, **65**(1), 96.
- 34 W. P. Zang, J. G. Tian, G. Y. Zhang, *et al.*, Theoretical analysis of Z-scan in thick optical nonlinear media, *Acta Phys. Sin.*, 1994, (3), 476–482.
- 35 A. Cetin, R. Kibar, M. Hatipoglu, *et al.*, Third-order optical nonlinearities of Cu and Tb nanoparticles in  $\text{SrTiO}_3$ , *Phys. B*, 2010, **405**(9), 2323–2325.
- 36 S. J. Wagner, B. M. Holmes, U. Younis, *et al.*, Controlling Third-Order Nonlinearities by Ion-Implantation Quantum-Well Intermixing, *IEEE Photonics Technol. Lett.*, 2009, **21**(2), 85–87.
- 37 M. Bayle, C. Bonafos, P. Benzo, *et al.*, Ag doped silicon nitride nanocomposites for embedded plasmonics, *Appl. Phys. Lett.*, 2015, **107**(10), 101907.
- 38 Y. H. Wang, S. J. Peng, J. D. Lu, *et al.*, Nonlinear optical properties of Ag nanocluster composite fabricated by 200 keV negative ion implantation, *Vacuum*, 2008, **83**(2), 412–415.
- 39 T. Cesca, G. Pellegrini, V. Bello, *et al.*, Nonlinear optical properties of Au-Ag nanoplanets made by ion beam processing of bimetallic nanoclusters in silica, *Nucl. Instrum. Methods Phys. Res.*, 2010, **268**(19), 3227–3230.
- 40 M. Song, Q. M. Liu, G. X. Cai, *et al.*, Enhancement of third-order optical nonlinearities in  $72\text{GeS}_2\text{-}18\text{Ga}_2\text{S}_3\text{-}10\text{CdS}$  glasses by Ag ion implantation, *Chalcogenide Lett.*, 2015, **12**(9), 453–460.
- 41 O. Sánchezdena, P. Motasantiago, L. Tamayorivera, *et al.*, Size-and shape-dependent nonlinear optical response of Au nanoparticles embedded in sapphire, *Opt. Mater. Express*, 2013, **4**(1), 92.
- 42 A. I. Rysanyanskiy, B. Palpant, S. Debrus, *et al.*, Third-order nonlinear-optical parameters of gold nanoparticles in different matrices, *J. Lumin.*, 2007, **127**(1), 181–185.
- 43 J. W. Chen, Q. Ren, X. Q. Wang, *et al.*, Third-Order Nonlinear Optical Properties of 1,3-Dithiole-2-thione-4,5-dithiolate Compounds at Different Wavelengths, *Acta Phys.-Chim. Sin.*, 2012, **28**(4), 942–948.
- 44 C. Torres-Torres, A. López-Suárez, B. Can-Uc, *et al.*, Collective optical Kerr effect exhibited by an integrated configuration of silicon quantum dots and gold nanoparticles embedded in ion-implanted silica, *Nanotechnology*, 2015, **26**(29), 295701.
- 45 A. Agrawal, T. Ahmad Dar, R. Solanki, *et al.*, Study of nonlinear optical properties of pure and Mg-doped ZnO films, *Phys. Status Solidi*, 2015, **252**(8), 1–6.
- 46 Z. Chai, Y. Zhi, L. Liu, *et al.*, Effect of absorption on the diffraction efficiency in  $\text{Fe:LiNbO}_3$  crystal with  $90^\circ$  recording geometry, *J. Opt. A: Pure Appl. Opt.*, 2005, **7**(12), 711.
- 47 A. Dakka, J. Lafait, C. Sella, *et al.*, Optical properties of Ag- $\text{TiO}_2$  nanocermet films prepared by cosputtering and multilayer deposition techniques, *Appl. Opt.*, 2000, **39**(16), 2745.
- 48 U. Pal, E. A. Almanza, O. V. Cuchillo, *et al.*, Preparation of Au/ZnO nanocomposites by radio frequency co-sputtering, *Sol. Energy Mater. Sol. Cells*, 2001, **70**(3), 363–368.



- 49 N. Pinçon, B. Palpant, D. Prot, *et al.*, Third-order nonlinear optical response of Au:SiO<sub>2</sub>, thin films: Influence of gold nanoparticle concentration and morphologic parameters, *Eur. Phys. J. D*, 2002, **19**(3), 395–402.
- 50 R. Adair, L. L. Chase and S. A. Payne, Nonlinear refractive index of optical crystals, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1989, **39**(5), 3337–3350.
- 51 X. J. Zhang, W. Ji and S. H. Tang, Determination of optical nonlinearities and carrier lifetime in ZnO, *J. Opt. Soc. Am. B*, 1997, **14**(8), 1951–1955.
- 52 B. Jalali, Silicon photonics: Nonlinear optics in the mid-infrared, *Nat. Photonics*, 2010, **4**(8), 506–508.
- 53 H. K. Tsang and Y. Liu, Nonlinear optical properties of silicon waveguides, *Semicond. Sci. Technol.*, 2008, **23**(23), 1129–1132.
- 54 H. Ji, M. Galili, H. Hu, *et al.*, 1.28-Tb/s Demultiplexing of an OTDM DPSK Data Signal Using a Silicon Waveguide, *IEEE Photonics Technol. Lett.*, 2010, **22**(23), 1762–1764.
- 55 D. A. B. Miller, Optical interconnects to electronic chips, *Appl. Opt.*, 2010, **49**(25), 59–70.
- 56 P. Zhang, X. Zhang, L. Peng, *et al.*, Interface state-related linear and nonlinear optical properties of nanocrystalline Si/SiO<sub>2</sub> multilayers, *Appl. Surf. Sci.*, 2014, **292**(1), 262–266.
- 57 P. Zhang, X. Zhang, J. Xu, *et al.*, Tunable nonlinear optical properties in nanocrystalline Si/SiO<sub>2</sub> multilayers under femtosecond excitation, *Nanoscale Res. Lett.*, 2014, **9**(1), 28.
- 58 Z. Bisadi, M. Mancinelli, S. Manna, *et al.*, *Phys. Status Solidi A*, 2016, **212**(12), 2659–2671.
- 59 W. Hou and S. B. Cronin, A Review of Surface Plasmon Resonance-Enhanced Photocatalysis, *Adv. Funct. Mater.*, 2013, **23**(13), 1612–1619.
- 60 F. Jiang, D. Chen, R. Li, *et al.*, Eco-friendly synthesis of size-controllable amine-functionalized graphene quantum dots with antimycoplasma properties, *Nanoscale*, 2013, **5**(3), 1137–1142.
- 61 T. J. Antosiewicz and S. P. Apell, Plasmonic glasses: optical properties of amorphous metal-dielectric composites, *Opt. Express*, 2014, **22**(2), 2031.
- 62 C. B. D. Araújo, L. R. P. Kassab, C. T. Dominguez, *et al.*, Photoluminescence and nonlinear optical phenomena in plasmonic random media—A review of recent works, *J. Lumin.*, 2016, **169**, 492–496.
- 63 G. Fan, S. Ren, S. Qu, *et al.*, Mechanisms for fabrications and nonlinear optical properties of Pd and Pt nanoparticles by femtosecond laser, *Opt. Commun.*, 2013, **295**(295), 219–225.
- 64 A. Chehrghani and M. J. Torkamany, Nonlinear optical properties of laser synthesized Pt nanoparticles: saturable and reverse saturable absorption, *Laser Phys.*, 2014, **24**(1), 015901.
- 65 H. J. Chen, Y. H. Wang, X. J. Zhang, *et al.*, Structure analysis of bimetallic Co–Au nanoparticles formed by sequential ion implantation, *Appl. Surf. Sci.*, 2016, **378**(378), 191–195.
- 66 B. Can-Uc, R. Rangel-Rojo, A. Peña-Ramírez, *et al.*, Nonlinear optical response of platinum nanoparticles and platinum ions embedded in sapphire, *Opt. Express*, 2016, **24**(9), 9955.
- 67 B. Can-Uc, R. Rangel-Rojo, H. Márquez, *et al.*, Nanoparticle containing channel waveguides produced by a multi-energy masked ion-implantation process, *Opt. Express*, 2015, **23**(3), 3176–3185.
- 68 R. Li, N. N. Dong, *et al.*, Giant Enhancement of Nonlinear Optical Response in Nd:YAG Single Crystals by Embedded Silver Nanoparticles, *ACS Omega*, 2017, **2**, 1279–1286.
- 69 Y. Hua, K. Chandra, D. H. M. Dam, *et al.*, Shape-Dependent Nonlinear Optical Properties of Anisotropic Gold Nanoparticles, *J. Phys. Chem. Lett.*, 2015, **6**(24), 4904–4908.
- 70 R. Sato, M. Ohnuma, K. Oyoshi, *et al.*, Experimental investigation of nonlinear optical properties of Ag nanoparticles: Effects of size quantization, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **90**(12), 125417.
- 71 E. H. M. Sakho, O. S. Oluwafemi, P. Sreekanth, *et al.*, Improved nonlinear optical and optical limiting properties in non-covalent functionalized reduced graphene oxide/silver nanoparticle (NF-RGO/Ag-NPs) hybrid, *Opt. Mater.*, 2016, **58**, 476–483.
- 72 Q. Wan, T. H. Wang and C. L. Lin, Third-order optical nonlinearity and negative photoconductivity of Ge nanocrystals in Al<sub>2</sub>O<sub>3</sub> dielectric, *Nanotechnology*, 2003, **14**(11), L15.
- 73 M. P. Brenner, D. Lohse and T. F. Dupont, Brenner *et al.* Reply, *Phys. Rev. Lett.*, 1998, **80**(16), 3668–3669.
- 74 G. Ledoux, J. Gong, F. Huisken, *et al.*, Photoluminescence of size-separated silicon nanocrystals: Confirmation of quantum confinement, *Appl. Phys. Lett.*, 2002, **80**(25), 4834–4836.
- 75 H. Garcia, R. Kalyanaraman and R. Sureshkumar, Nonlinear optical properties of multi-metal nanocomposites in a glass matrix, *J. Phys. B: At., Mol. Opt. Phys.*, 2009, **42**(17), 175401.
- 76 Y. H. Wang, Y. M. Wang, J. D. Lu, *et al.*, Nonlinear optical properties of Cu nanoclusters by ion implantation in silicate glass, *Opt. Commun.*, 2010, **283**(3), 486–489.
- 77 B. Ghosh, P. Chakraborty, S. Mohapatra, *et al.*, Linear and nonlinear optical absorption in copper nanocluster-glass composites, *Mater. Lett.*, 2007, **61**(23–24), 4512–4515.
- 78 D. R. Symes, U. Wegner, H. C. Ahlswede, *et al.*, Ultrafast gated imaging of laser produced plasmas using the optical Kerr effect, *Appl. Phys. Lett.*, 2010, **96**(1), 755.
- 79 W. Tan, Y. Yang, J. Si, *et al.*, Shape measurement of objects using an ultrafast optical Kerr gate of bismuth glass, *J. Appl. Phys.*, 2010, **107**(4), 10.
- 80 G. Mattei, P. Mazzoldi and H. Bernas, Metal Nanoclusters for Optical Properties, *Top. Appl. Phys.*, 2009, **116**, 287–316.
- 81 K. Uchida, S. Kaneko, S. Omi, *et al.*, Optical nonlinearities of a high concentration of small metal particles dispersed in glass: copper and silver particles, *J. Opt. Soc. Am. B*, 1994, **11**(7), 1236–1243.
- 82 J. Qiu, X. Jiang, C. Zhu, *et al.*, Manipulation of gold nanoparticles inside transparent materials, *Angew. Chem.*, 2004, **43**(17), 2230–2234.
- 83 G. Lin, D. Tan, F. Luo, *et al.*, Linear and nonlinear optical properties of glasses doped with Bi nanoparticles, *J. Non-Cryst. Solids*, 2011, **357**(11), 2312–2315.



- 84 N. Venkatram, R. Sathyavathi and D. N. Rao, Size dependent multiphoton absorption and refraction of CdSe nanoparticles, *Opt. Express*, 2007, **15**(19), 12258–12263.
- 85 M. S. Neo, N. Venkatram, G. S. Li, *et al.*, Size-Dependent Optical Nonlinearities and Scattering Properties of PbS Nanoparticles, *J. Phys. Chem. C*, 2015, **113**(44), 19055.
- 86 L. Irimpan, V. P. N. Nampoori, P. Radhakrishnan, *et al.*, Size-dependent enhancement of nonlinear optical properties in nanocolloids of ZnO, *J. Appl. Phys.*, 2008, **103**(3), 9797.
- 87 H. S. Jun, K. S. Lee, S. H. Yoon, *et al.*, 3rd order nonlinear optical properties of Au:SiO<sub>2</sub>, nanocomposite films with varying Au particle size, *Phys. Status Solidi A*, 2010, **203**(6), 1211–1216.
- 88 J. C. Hulthen and R. P. Van Duyne, Nanosphere lithography: A materials general fabrication process for periodic particle array surfaces, *J. Vac. Sci. Technol., A*, 1995, **13**(3), 1553–1558.
- 89 E. H. M. Sakho, O. S. Oluwafemi, P. Sreekanth, *et al.*, Improved nonlinear optical and optical limiting properties in non-covalent functionalized reduced graphene oxide/silver nanoparticle (NF-RGO/Ag-NPs) hybrid, *Opt. Mater.*, 2016, **58**, 476–483.
- 90 H. Pan, W. Chen, Y. P. Feng, *et al.*, Optical limiting properties of metal nanowires, *Appl. Phys. Lett.*, 2006, **88**(22), 1316.
- 91 C. S. S. Sandeep, A. K. Samal, T. Pradeep, *et al.*, Optical limiting properties of Te and Ag<sub>2</sub>Te nanowires, *Chem. Phys. Lett.*, 2010, **485**(4), 326–330.
- 92 E. M. Garmire, Photonics: Linear and Nonlinear Interactions of Laser Light and Matter, *Phys. Today*, 2002, **55**(3), 68.
- 93 K. Padmanabhan, S. E. Andrews and J. A. Fitzpatrick, *Multiphoton imaging*, 2010, ch. 2.
- 94 T. Huang, Z. Hao, H. Gong, *et al.*, Third-order nonlinear optical properties of a new copper coordination compound: A promising candidate for all-optical switching, *Chem. Phys. Lett.*, 2008, **451**(4), 213–217.
- 95 Y. Takeda, O. A. Plaksin, J. Lu, *et al.*, Optical switching performance of metal nanoparticles fabricated by negative ion implantation, *Nucl. Instrum. Methods Phys. Res.*, 2006, **242**(1–2), 194–197.
- 96 R. Huschka, J. Zuloaga, M. W. Knight, *et al.*, Light-induced release of DNA from gold nanoparticles: nanoshells and nanorods, *J. Am. Chem. Soc.*, 2011, **133**(31), 12247–12255.
- 97 R. Bardhan, S. Lal, A. Joshi, *et al.*, Theranostic Nanoshells: From Probe Design to Imaging and Treatment of Cancer, *Acc. Chem. Res.*, 2011, **44**(10), 936–946.
- 98 D. Denkova, Imaging the Magnetic Near-Field of Plasmon Modes in Bar Antennas, *Optical Characterization of Plasmonic Nanostructures: Near-Field Imaging of the Magnetic Field of Light*, Springer International Publishing, 2016.
- 99 G. D. Hale, J. B. Jackson, O. E. Shmakova, *et al.*, Enhancing the active lifetime of luminescent semiconducting polymers via doping with metal nanoshells, *Appl. Phys. Lett.*, 2001, **78**(11), 1502–1504.
- 100 R. Bardhan, N. K. Grady, T. Ali, *et al.*, Metallic nanoshells with semiconductor cores: optical characteristics modified by core medium properties., *ACS Nano*, 2010, **4**(10), 6169–6179.
- 101 A. Salah, A. Mansour, M. B. Mohamed, *et al.*, Effects of nanoparticles size and concentration and laser power on nonlinear optical properties of Au and Au–CdSe nanocrystals, *Appl. Surf. Sci.*, 2015, **353**, 112–117.
- 102 Q. Wang, L. Song, J. Qiao, *et al.*, CdS–CdSe (CdTe) core-shell quantum dots sensitized TiO<sub>2</sub>, nanotube array solar cells, *Sol. Energy Mater. Sol. Cells*, 2015, **132**, 650–654.
- 103 M. S. Al-Kotb, J. Z. Al-Waheidi and M. F. Kotkata, Opto-electronic characterizations of oriented nano-structure CdSe film/Si (0 0 1) heterostructure, *Superlattices Microstruct.*, 2014, **69**(5), 149–163.
- 104 X. Yao, P. Yan, K. Zhang, *et al.*, Preparation of water-soluble CdSe quantum dots and its application for nitrite detection in the anodic electrochemiluminescence, *Luminescence*, 2013, **28**(4), 551–556.
- 105 S. Zhong, F. Zhang, B. Yu, *et al.*, Synthesis of PVP–Bi<sub>2</sub>WO<sub>6</sub>, photocatalyst and degradation of tetracycline hydrochloride under visible light, *J. Mater. Sci.: Mater. Electron.*, 2016, **27**(3), 3011–3020.
- 106 K. Salini, K. S. Rahul and V. Mathew, Size-dependent electronic and optical properties of an exciton in CdSe/CdS/CdSe/CdS multilayer spherical quantum dot, *Appl. Phys. A: Mater. Sci. Process.*, 2014, **116**(3), 1371–1377.
- 107 G. Kaur and S. K. Tripathi, Investigation of trypsin–CdSe quantum dot interactions via spectroscopic methods and effects on enzymatic activity, *Spectrochim. Acta, Part A*, 2015, **134**, 173–183.
- 108 S. K. Tripathi, R. Kaur, J. Kaur, *et al.*, Third-order nonlinear optical response of Ag–CdSe/PVA hybrid nanocomposite, *Appl. Phys. A: Mater. Sci. Process.*, 2015, **120**(3), 1047–1057.
- 109 A. Lyashchova, D. Fedorenko, Y. Garbovskiy, *et al.*, Strong thermal optical nonlinearity caused by CdSe nanoparticles synthesised in smectic ionic liquid crystal, *Liq. Cryst.*, 2013, **40**(10), 1377–1382.
- 110 K. M. Rahulan, G. Vinitha, S. Ganesan, *et al.*, Synthesis and optical limiting studies of HA and HA@SiO core-shell nanoparticles, *J. Mater. Sci.: Mater. Electron.*, 2012, **23**(4), 877–881.
- 111 Y. H. Wang, X. X. Yu, F. Liu, *et al.*, Nonlinear refraction of lithium niobate crystal doped with different metal nanoparticles, *Mater. Lett.*, 2014, **123**(9), 35–37.

