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# Exploring the stable structures of cerium oxide nanoclusters using high-dimensional neural network potential†

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Cerium clusters have been extensively applied in industry owing to their extraordinary properties for oxygen storage and redox catalytic activities. However, their atomically precise structures have not been studied because of the lack of a reliable method to efficiently sample their complex structures. Herein, we combined a neural network algorithm with density functional theory calculations to establish a high-dimensional potential to search for the global minimums of cerium oxide clusters. Using  $\text{Ce}_{14}\text{O}_{28}$  as well as its reduced state  $\text{Ce}_{14}\text{O}_{27}$  and oxidized state  $\text{Ce}_{14}\text{O}_{29}$  with ultra-small dimensions of  $\sim 1.0$  nm as examples, we found that these three clusters adopt pyramid-like structures with the lowest energies, which was obtained by exploring 100 000 configurations in large feasible spaces. Further the neural network potential-enhanced molecular dynamics calculations indicated that these cluster structures are stable at high temperature. The electronic structure analysis suggested that these clusters are highly active and easily lose oxygen. This work demonstrated that neural network potentials can be useful for exploring the stable structures of metal oxide nanoclusters in practical applications.

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## Introduction

With unique redox properties, which are crucial for many applications,  $\text{CeO}_2$ -based materials have been widely used in fuel cells, water splitting, water-gas shift reactions, and others.<sup>1–9</sup> The unique redox properties of cerium oxides are mainly related to oxygen vacancies and  $\text{Ce}^{3+}$  ions.<sup>10–12</sup> In the past decade, nanoscale cerium oxides have received more attention than bulk materials because of their higher reactivity.<sup>13,14</sup> The high reactivity of ultra-small cerium oxides even at ambient temperatures gives them many potential applications, especially in biomedical fields, such as antibacterial agents, cancer therapies, and antioxidative neurorestoration.<sup>15–18</sup> The application of cerium oxide nanoparticles is related to their activity and stability, which depends on their size, shape, composition (such as  $\text{Ce}^{3+}/\text{Ce}^{4+}$  ratios), and other structural factors.<sup>19,20</sup> Obtaining atomically precise structures of cerium oxide nanoclusters can

help understand how structural factors affect their properties; however, identifying the structures of pristine  $\text{CeO}_2$  clusters is still challenging in experimental work.

Meanwhile, theoretical work has made great progress in exploring the stable structures of nanoclusters by combining the global-optimization method and first-principles calculations.<sup>21–24</sup> However, predicting the stable structures of metal oxides is more complicated as their possible configurations are numerous and the computational costs increase exponentially. Although there are some empirical potentials developed to aid approximates, such as the Lennard-Jones potential for  $\text{TiO}_2$  nanoparticles,<sup>25</sup> the modified embedded-atom method for ternary Cu-Ta-O oxides,<sup>26</sup> and the Morse function for  $\text{IrO}_2$  nanoclusters,<sup>27</sup> the complexity of the electronic configurations of  $\text{CeO}_2$  clusters requires a high-precision potential using first-principles calculations for efficient structural sampling, which to date is still lacking.

In this work, a machine learning model was developed to construct the high-dimensional neural network potential to search for the most stable cerium oxide nanoclusters in large feasible spaces. The data set for the machine learning model included the coordinates and energies corresponding to geometry optimization of each random initial structure using density functional theory calculations. Using a well-trained neural network, the most stable structure was obtained by the basin-hopping global optimization of randomly generated isomers. Active learning was used in this workflow, iterating many times until the predicted energies fitted well with DFT

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calculations, not only the numerical values but also the order of the isomers. Using  $\text{Ce}_{14}\text{O}_{28}$  as well as its reduced state  $\text{Ce}_{14}\text{O}_{27}$  and oxidized state  $\text{Ce}_{14}\text{O}_{29}$  as examples, we found three clusters adopt pyramid-like structures as the global minima.

## Methods

### Generation of random initial structures

All the cerium oxide structures were randomly extracted from a large cerium oxide bulk structure. As shown in Fig. 1, taking the sampling gray region required to generate nanocluster  $\text{Ce}_{14}\text{O}_{28}$  as an example, a cerium atom was first selected as the center of the sampling region, and the 8 oxygen atoms closest to it were preferentially selected (the atomic distances to the central atom were 2.37 Å), and then 20 oxygen atoms were randomly selected from the second nearest neighboring 24 O atoms with atomic distances of 4.53 Å. The remaining 13 Ce atoms were randomly selected from a total of 18 cerium atoms of the nearest neighbor with atomic distances of 3.86 Å and the second nearest neighbor with atomic distances of 5.46 Å. For securing the randomness of the constructed structures, all the atoms were given random displacements of no more than 0.2 Å compared to the initial positions.

### Data set based on DFT calculations

The data set of the machine learning model was based on density functional theory (DFT) calculations, performed using the Vienna *ab initio* Simulation Package (VASP).<sup>28</sup> All the DFT calculations employed the general gradient approximation (GGA) Perdew–Burke–Ernzerhof (PBE) functional.<sup>29</sup> The cutoff energy of the plane-wave basis was set to 400 eV. The energies were converged to  $10^{-4}$  eV. The conjugate gradient algorithm was employed for geometry optimization. The DFT + U method<sup>30</sup> was used in these calculations with an effective U of 5.0 eV. Considering the rationality of the initial structures, the atomic spacings were calculated to determine whether there were outlier atoms. If outlier atoms existed, the random structure was re-generated. These structures were optimized by DFT calculations, and the coordinates and energies were used as the data set for machine learning. The total data set was randomly split into 90% for training, while the remaining 10% was used as the test data set to validate the accuracy of the machine learning potential.

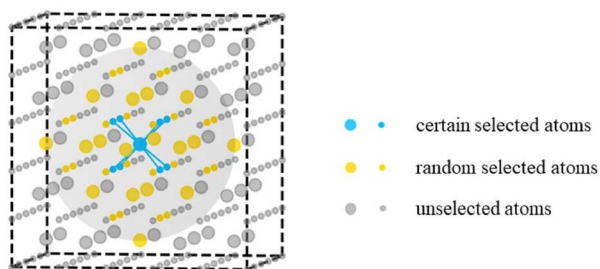


Fig. 1 Schematic diagram of the sampling method. The larger atoms represent cerium atoms and the smaller atoms represent oxygen atoms, while the gray sphere shows the sampling area.

### Construction of the machine learning model

The machine learning model of a class of high-dimensional neural network, based on the Behler–Parrinello artificial neural network, was constructed using the n2p2 code.<sup>31–33</sup> Using this code, the total potential energy of the cerium oxide was defined as the sum of atomic potential energies. The atomic energies were trained by an atomic neural network using the symmetry function to represent the structural and chemical environment information of each atom. The symmetry functions, including radial symmetry function, narrow angular symmetry function, and wide angular symmetry function, were chosen to represent the local chemical environment of cerium atoms and oxygen atoms from the Cartesian coordinates. The cutoff radius of all the symmetry functions was 6 Å. Each neural network consisted of an input layer, two hidden layers with 18 nodes each, and an output layer. The weight parameters of the neural network were optimized by Kalman filter,<sup>34</sup> which is an efficient approach to minimize the cost function defined by the root-mean-square error (RMSE) of energies and forces. In this machine learning model, active learning was used to iterate several times until the RMSE between the energies predicted by the neural network and the energies calculated by DFT converged to a value less than 10 meV per atom. More details about the machine learning method can be found in the ESI.†

### Global minimization based on the neural network potential

The neural network potential (NNP) combined with basin-hopping method was employed to explore the stable structures of the cerium oxide nanoclusters, using the large-scale atomic/molecular massively parallel simulator (LAMMPS).<sup>35</sup> For sampling larger configuration spaces, we adopted another sampling strategy for global minimization, in which all the atoms in the sampling region were selected with the same probability, and the central atom of the sampling region was iteratively updated. All the random structures were generated by this method with different random seeds for each structure. The energy of each structure was minimized using the conjugate gradient algorithm based on the neural network potential function, setting  $10^{-5}$  eV as the energy tolerance. A total of 10 paths were run in parallel to search for the most stable structure of cerium oxide nanoclusters in a huge configuration space, with 10 000 structures per path. Combining the results of these 10 paths, the lowest energy nanocluster after geometry optimization was considered as the most stable structure.

## Results and discussion

The machine learning model started with generating a total of ~300 non-repeating initial structures of the three types of cerium oxides ( $\text{Ce}_{14}\text{O}_{27}$ ,  $\text{Ce}_{14}\text{O}_{28}$ ,  $\text{Ce}_{14}\text{O}_{29}$ ) with the size of ~1 nm, which were randomly sampled by the number of cerium and oxygen atoms in a large cerium oxide crystal structure. After optimizing the random initial structures by DFT calculations, the structures and energies were used as the data set for the machine learning model. In this work, active learning was used and the three neural networks were iterated several times until



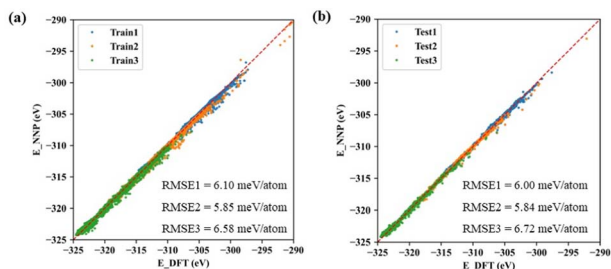


Fig. 2 Comparison of the energies calculated using DFT and predicted using neural network potential (NNP): (a) training; (b) testing (1: neural network of  $Ce_{14}O_{27}$ , 2: neural network of  $Ce_{14}O_{28}$ , and 3: neural network of  $Ce_{14}O_{29}$ ).

the predicted results were in good agreement with the values and ordering of the energies calculated by the DFT, increasing the data points to 8142, 7796, and 10 385. Finally, the root-mean-square error (RMSE) of acceptable neural networks was about 6 meV per atom (Fig. 2).

The predictions for the top 20 low-energy structures of the NNP were in good agreement with DFT calculations, as shown in Fig. 3. The mean absolute energy errors of these top 20 lowest energy nanoclusters  $Ce_{14}O_x$  ( $x = 27, 28,$  and  $29$ ) were 0.36, 0.15, and 0.41 eV, and the mean errors to each atom were 8.8, 3.6, and 9.5 meV per atom, all of which were less than 10 meV per atom. For each type of cerium oxide nanocluster, the top two minima predicted by the NNP were fully consistent with the top two minima calculated by DFT. Although there were some fluctuations in the energy differences between NNP and DFT, especially for the nanocluster  $Ce_{14}O_{29}$ , the trend of energy ordering predicted by the NNP was similar to that calculated by DFT.

The top 3 minima obtained from the energy verified by DFT calculations of the nanoclusters  $Ce_{14}O_x$  ( $x = 27, 28,$  and  $29$ ) are shown in Fig. 4, and the others are shown in Fig. S1–S3.† All the most stable structures had higher symmetry and lower energies by at least 0.6 eV than the others. These structures were all pyramid-like, where the cerium atoms were in the same configuration and oxygen atoms were nearly the same

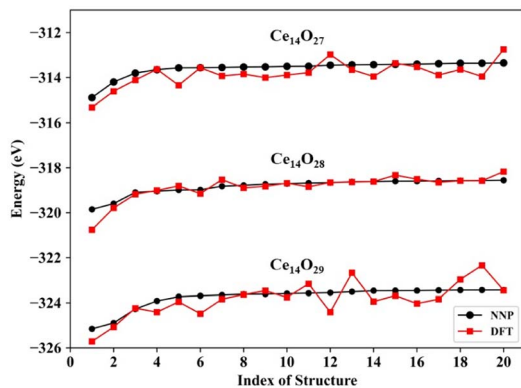


Fig. 3 Comparison of the NNP and DFT energies of the top 20 low-energy structures of the  $Ce_{14}O_x$  ( $x = 27, 28,$  and  $29$ ) structures (NNP: energies by neural network potential prediction, DFT: energies by DFT calculation).

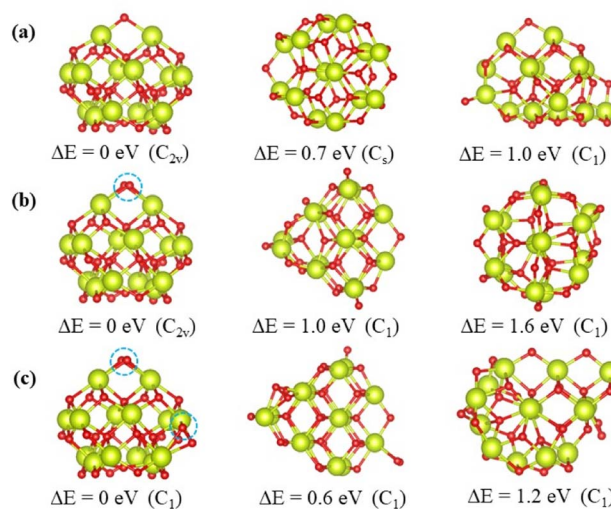


Fig. 4 Most stable isomer and two other less stable ones, including the corresponding structure, symmetry, and energy relative to the most stable structure: (a)  $Ce_{14}O_{27}$ ; (b)  $Ce_{14}O_{28}$ ; (c)  $Ce_{14}O_{29}$  (red: oxygen atoms and green: cerium atoms). Circled in blue are locations where the most stable structures differ from the most stable structure of the nanocluster  $Ce_{14}O_{27}$ .

configuration. The 14 cerium atoms consisted of three Ce layers in a  $2 \times 1 : 3 \times 2 : 2 \times 3$  pattern. The oxygen atoms consisted of four O layers in slightly different pattern. As shown in Fig. 4, the oxygen atoms circled in blue are the additional oxygen atoms of  $Ce_{14}O_{28}$  and  $Ce_{14}O_{29}$  compared to  $Ce_{14}O_{27}$ . The most stable structure of the nanocluster  $Ce_{14}O_{28}$  was based on the structure of  $Ce_{14}O_{27}$ , adding a new oxygen atom next to the oxygen atom in the top layer and maintaining the same  $C_{2v}$  symmetry as the nanocluster  $Ce_{14}O_{27}$ . As the most stable structure of the nanocluster  $Ce_{14}O_{29}$ , a new oxygen atom was added to the right corner of the third layer of oxygen atoms on the structure of  $Ce_{14}O_{28}$ , but the symmetry was reduced to  $C_1$ .

The lowest oxygen vacancy formation energies of  $CeO_2(111)$  and bulk  $CeO_2$  calculated by DFT were 1.63 and 3.30 eV. In comparison, the energies for the nanoclusters  $Ce_{14}O_{29}$  and  $Ce_{14}O_{28}$  to lose one oxygen atom were 0.03 and 0.15 eV, respectively, which are significantly lower than those of  $CeO_2(111)$  and bulk  $CeO_2$ , indicating cerium oxide nanoclusters are more likely to store and lose oxygen atoms for redox reactions.

Based on neural network potential, molecular dynamics (MD) simulations of the three most stable structures of the nanoclusters  $Ce_{14}O_x$  ( $x = 27, 28,$  and  $29$ ) were performed for up to 1 nanosecond (ns) at 700 K to verify the dynamic stability. The simulations were performed in the *NVT* ensemble with time steps of 1 femtosecond (fs). The energies of  $Ce_{14}O_x$  ( $x = 27, 28,$  and  $29$ ) fluctuated around  $-311.2$ ,  $-316.0$ , and  $-320.9$  eV, respectively. As shown in Fig. 5, the snapshots taken every 200 picosecond (ps) show that the structure of  $Ce_{14}O_{28}$  was more stable over the simulation period of 1 ns, with almost no significant changes. Similar structural stabilities could also be found in  $Ce_{14}O_{27}$  and  $Ce_{14}O_{29}$  (Fig. S5 and S6†). Compared to



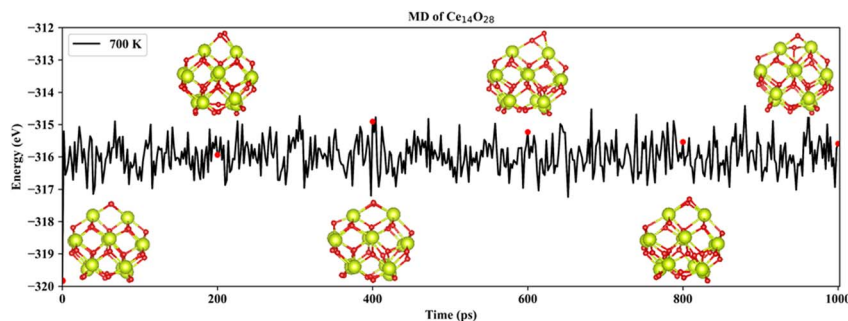


Fig. 5 Snapshots of the most stable nanocluster  $\text{Ce}_{14}\text{O}_{28}$  extracted every 200 ps from 700 K MD trajectories.

the nanocluster  $\text{Ce}_{14}\text{O}_{27}$ , the nanoclusters  $\text{Ce}_{14}\text{O}_{28}$  and  $\text{Ce}_{14}\text{O}_{29}$  were more active, especially  $\text{Ce}_{14}\text{O}_{29}$ . These structural changes occur at the locations of the additional oxygen atoms compared to the nanocluster  $\text{Ce}_{14}\text{O}_{27}$ , where the cerium–oxygen bonds briefly break and then return to the previous structure, indicating that these locations may be the active sites of the structures.

Compared to the bulk structure of  $\text{CeO}_2$ , which is composed of 8-coordinate  $\text{Ce}^{4+}$  and 4-coordinate  $\text{O}^{2-}$  ions, the Ce atoms in the three most stable nanoclusters  $\text{Ce}_{14}\text{O}_x$  ( $x = 27, 28$ , and  $29$ ) are 5-coordinate (green) or 6-coordinate (cyan), as shown in Fig. 6(a).  $\text{Ce}_{14}\text{O}_{27}$  has four 6-coordinate cerium atoms, while  $\text{Ce}_{14}\text{O}_{28}$  and  $\text{Ce}_{14}\text{O}_{29}$  have six and eight 6-coordinate cerium atoms, respectively. The oxygen atoms are 2, 3, and 4 coordinated. The inner oxygen atoms of the cerium oxide cluster are 4-coordinate and the outer oxygen atoms are 3- or 2-coordinate. The three most stable nanoclusters  $\text{Ce}_{14}\text{O}_x$  ( $x = 27, 28$ , and  $29$ ) all have five 4-coordinate oxygen atoms. For each of most stable nanoclusters, all the 4-coordinate oxygen atoms are in the same position, where one is in the middle of the second layer of

oxygen atoms, and the other four atoms are symmetrically located inside the third layer in a  $2 \times 2$  pattern.

As shown in Fig. 6(a), the locations of  $\text{Ce}^{3+}$  in the three lowest energy structures were characterized according to the spin density. There are two  $\text{Ce}^{3+}$  in the most stable structures of each nanocluster  $\text{Ce}_{14}\text{O}_x$  ( $x = 27, 28$ , and  $29$ ). The two  $\text{Ce}^{3+}$  in the nanoclusters  $\text{Ce}_{14}\text{O}_{28}$  and  $\text{Ce}_{14}\text{O}_{29}$  are located at the same location, symmetrically within the first layer of cerium atoms. Two  $\text{Ce}^{3+}$  of the nanocluster  $\text{Ce}_{14}\text{O}_{27}$  are located on the right side of the first layer of cerium atoms and on the left of the second layer. Since the volume of  $\text{Ce}^{3+}$  is larger than that of  $\text{Ce}^{4+}$ , the asymmetrical location of the two  $\text{Ce}^{3+}$  distorts the structure and affects the symmetry of the nanocluster  $\text{Ce}_{14}\text{O}_{27}$ . The symmetry of the nanocluster  $\text{Ce}_{14}\text{O}_{27}$  was reduced from  $C_{2v}$  to  $C_s$ . Interestingly, both the most stable structures of  $\text{Ce}_{14}\text{O}_{28}$  and  $\text{Ce}_{14}\text{O}_{29}$  have two  $\text{Ce}^{3+}$  polarons, which are different from the bulk and surface of  $\text{CeO}_2$ . This could be attributed to their distinct structures, in which each additional oxygen atom of the  $\text{Ce}_{14}\text{O}_{28}$  and  $\text{Ce}_{14}\text{O}_{29}$  forms a  $1.5 \text{ \AA}$  O–O bond with the adjacent oxygen atom. Thus, the O–O species contribute two electrons to the  $\text{Ce}_{14}\text{O}_{28}$  and  $\text{Ce}_{14}\text{O}_{29}$  as the O of  $\text{Ce}_{14}\text{O}_{27}$ . This means that all these three clusters have electrons they can easily lose, indicating their high catalytic activity.

The Ce–O bond length in the crystal structure of cerium oxide is  $2.34 \text{ \AA}$ . The average bond lengths in the global minima of the nanoclusters  $\text{Ce}_{14}\text{O}_x$  ( $x = 27, 28$ , and  $29$ ) are  $2.28, 2.28$ , and  $2.29 \text{ \AA}$ , respectively, which are shorter than the Ce–O bond in the crystal structure. As shown in Fig. 6(b) for the statistics of the bond lengths at  $0.1 \text{ \AA}$  intervals, the Ce–O bond lengths of the three most stable cerium oxide nanoclusters are  $2.3 \pm 0.3 \text{ \AA}$ , and about 60% of the bond lengths are in the range of  $2.3 \pm 0.1 \text{ \AA}$ , with more than half less than  $2.3 \text{ \AA}$ . Starting from the nanocluster  $\text{Ce}_{14}\text{O}_{27}$  with 74 Ce–O bonds, two Ce–O bonds are added for each additional oxygen atom. The longest Ce–O bond lengths of these three most stable structures are  $2.52, 2.46$ , and  $2.47 \text{ \AA}$ . The longest bond length of the nanocluster  $\text{Ce}_{14}\text{O}_{27}$  is significantly longer than the other two stable structures, and is  $0.04 \text{ \AA}$  longer than the second longest bond of its own. Meanwhile, the differences between the longest bond length and the second bond length of  $\text{Ce}_{14}\text{O}_{28}$  or  $\text{Ce}_{14}\text{O}_{29}$  does not exceed  $0.01 \text{ \AA}$ . In  $\text{Ce}_{14}\text{O}_{27}$ , the longest bond length is between the  $\text{Ce}^{3+}$  in the second layer of cerium atom and the leftmost oxygen atom in the second layer of oxygen atoms. This  $\text{Ce}^{3+}$  in  $\text{Ce}_{14}\text{O}_{27}$  has longer

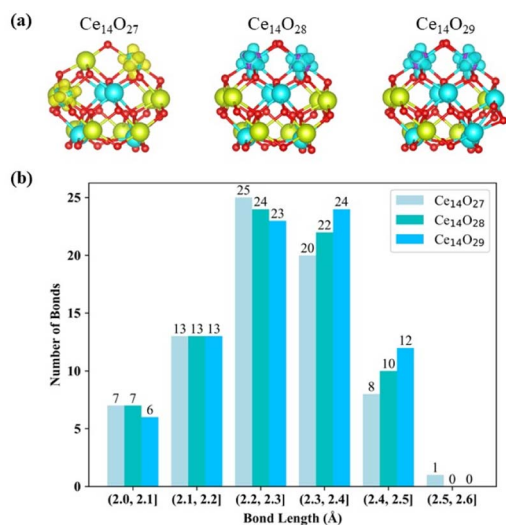


Fig. 6 (a) Structures of cerium oxides (red: oxygen atoms, green: 5-coordinate cerium atoms, and cyan: 6-coordinate cerium atoms). (b) Statistics of the bond lengths in the most stable structures of  $\text{Ce}_{14}\text{O}_x$  ( $x = 27, 28$ , and  $29$ ).



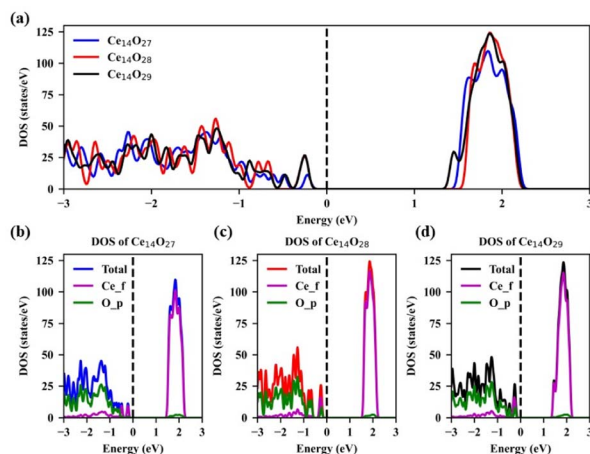


Fig. 7 Density of states (DOS) of the most stable structures of  $\text{Ce}_{14}\text{O}_x$  ( $x = 27, 28, \text{ and } 29$ ): (a) TDOS of  $\text{Ce}_{14}\text{O}_x$  ( $x = 27, 28, \text{ and } 29$ ); (b) DOS of  $\text{Ce}_{14}\text{O}_{27}$ ; (c) DOS of  $\text{Ce}_{14}\text{O}_{28}$ ; (d) DOS of  $\text{Ce}_{14}\text{O}_{29}$ .

bond lengths with the surrounding oxygen atoms compared to the  $\text{Ce}^{4+}$  atom at the same location in the nanocluster  $\text{Ce}_{14}\text{O}_{28}$  or  $\text{Ce}_{14}\text{O}_{29}$ . It thus seems that  $\text{Ce}^{3+}$  is larger and requires more space, so it pulls away the surrounding oxygen atoms. The left-most oxygen atom in the second layer is more likely to be pulled away, so the bond length between this oxygen and  $\text{Ce}^{3+}$  in the second layer of cerium atoms is significantly longer.

The density of states (DOS) of the most stable structures of the nanoclusters  $\text{Ce}_{14}\text{O}_x$  ( $x = 27, 28, \text{ and } 29$ ) are shown in Fig. 7. The Fermi levels of these three structures are close to the valence band maximum (VBM) and the corresponding DOSs are zero. The band gaps are about 1.38, 1.48, and 1.30 eV for  $\text{Ce}_{14}\text{O}_{27}$ ,  $\text{Ce}_{14}\text{O}_{28}$ , and  $\text{Ce}_{14}\text{O}_{29}$ , respectively. In the conduction band, the DOS value of the nanoclusters  $\text{Ce}_{14}\text{O}_{27}$  is slightly lower than that of the nanoclusters  $\text{Ce}_{14}\text{O}_{28}$  and  $\text{Ce}_{14}\text{O}_{29}$ , while the DOS values of the nanoclusters  $\text{Ce}_{14}\text{O}_{28}$  and  $\text{Ce}_{14}\text{O}_{29}$  are almost equal. All the VBM are mainly contributed by the p orbital of O, while the conduction band minimum (CBM) are contributed mainly by the f orbital of Ce.

## Conclusions

In conclusion, a machine learning model was developed to predict the most stable structures of cerium oxide nanoclusters. Using the well-trained high-dimensional neural network potential combined with the basin-hopping method, the global minimum structures of  $\text{Ce}_{14}\text{O}_{28}$  as well as its reduced state  $\text{Ce}_{14}\text{O}_{27}$  and oxidized state  $\text{Ce}_{14}\text{O}_{29}$  were explored. The  $\text{Ce}_{14}\text{O}_x$  ( $x = 27, 28, \text{ and } 29$ ) clusters adopted similar pyramid-like structures. The NNP-MD simulation suggested these cluster structures are stable at high temperature. The electronic structures analysis indicated these clusters are highly active and can easily lose oxygen, which leads to their resultant antioxidant activity. Moreover, due to the stability of these ultra-small nanoclusters, they will be excellent antioxidant agents in biomedical fields for the treatment of oxygen-related diseases. This work provides a viable strategy to search for the stable structures of cerium

oxides nanoclusters. In addition, using the global optimization combined with the neural network potential, this method has both the advantages of high efficiency and high accuracy compared to first-principles calculations and the empirical potentials, which indicates its potential for exploring more complex systems in the future.

## Author contributions

Y. G. initiated the project. Y. G. and Q. R. supervised the project. H. C. carried out the calculations and data analysis. H. C. drafted the paper. Y. G. revised the paper. All authors participated in the discussions.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

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## References

- 1 T. Montini, M. Melchionna, M. Monai and P. Fornasiero, *Chem. Rev.*, 2016, **116**, 5987–6041.
- 2 R. Fiala, A. Figueroba, A. Bruix, M. Vaclavu, A. Rednyk, I. Khalakhan, M. Vorokhta, J. Lavkova, F. Illas, V. Potin, I. Matolinova, K. M. Neyman and V. Matolin, *Appl. Catal., B*, 2016, **197**, 262–270.
- 3 M. Riegraf, I. Bombarda, F. Dömling, T. Liensdorf, C. Sitzmann, N. Langhof, S. Schafföner, F. Han, N. Sata, C. Geipel, C. Walter and R. Costa, *ACS Appl. Mater. Interfaces*, 2021, **13**, 49879–49889.
- 4 S. Wen, J. Huang, T. Li, W. Chen, G. Chen, Q. Zhang, X. Zhang, Q. Qian and K. Ostrikov, *Appl. Catal., B*, 2022, **316**, 121678.
- 5 R. Li, C. Wen, K. Yan, T. Liu, B. Zhang, M. Xu and Z. Zhou, *J. Mater. Chem. A*, 2023, **11**, 7128–7141.
- 6 J.-O. Shim, H.-S. Na, A. Jha, W.-J. Jang, D.-W. Jeong, I. W. Nah, B.-H. Jeon and H.-S. Roh, *Chem. Eng. J.*, 2016, **306**, 908–915.



- 7 J. Lee, D. Shin, E. Lee, C. Li, J. M. Kim, J. W. Han and D. H. Kim, *Appl. Catal., B*, 2022, **305**, 121038.
- 8 W. Yang, Z. Wang, Y. Wei, Y. Xia, Z. Zhu and C. Liu, *Chem. Eng. J.*, 2022, **446**, 136740.
- 9 D. Van Dao, H. D. Jung, T. T. D. Nguyen, S.-W. Ki, H. Son, K.-B. Bae, T. D. Le, Y.-H. Cho, J.-K. Yang, Y.-T. Yu, S. Back and I.-H. Lee, *J. Mater. Chem. A*, 2021, **9**, 10217–10230.
- 10 Z.-K. Han, Y.-Z. Yang, B. Zhu, M. V. Ganduglia-Pirovano and Y. Gao, *Phys. Rev. Mater.*, 2018, **2**, 035802.
- 11 D. Zhang, Z.-K. Han, G. E. Murgida, M. V. Ganduglia-Pirovano and Y. Gao, *Phys. Rev. Lett.*, 2019, **122**, 096101.
- 12 F. Esch, S. Fabris, L. Zhou, T. Montini, C. Africh, P. Fornasiero, G. Comelli and R. Rosei, *Science*, 2005, **309**, 752–755.
- 13 C. Sun, H. Li and L. Chen, *Energy Environ. Sci.*, 2012, **5**, 8475–8505.
- 14 S. D. Senanayake, D. Stacchiola and J. A. Rodriguez, *Acc. Chem. Res.*, 2013, **46**, 1702–1711.
- 15 M. A. Saifi, S. Seal and C. Godugu, *J. Controlled Release*, 2021, **338**, 164–189.
- 16 E. Casals, M. Zeng, M. Parra-Robert, G. Fernández-Varo, M. Morales-Ruiz, W. Jiménez, V. Puentes and G. Casals, *Small*, 2020, **16**, 1907322.
- 17 J. Kim, G. Hong, L. Mazaleuskaya, J. C. Hsu, D. N. Rosario-Berrios, T. Grosser, P. F. Cho-Park and D. P. Cormode, *ACS Appl. Mater. Interfaces*, 2021, **13**, 60852–60864.
- 18 X. Li, Z. Han, T. Wang, C. Ma, H. Li, H. Lei, Y. Yang, Y. Wang, Z. Pei, Z. Liu, L. Cheng and G. Chen, *Biomaterials*, 2022, **291**, 121904.
- 19 S. Yadav, S. Chamoli, P. Kumar and P. K. Maurya, *Int. J. Biol. Macromol.*, 2023, **246**, 125673.
- 20 S. Kargozar, F. Baino, S. J. Hoseini, S. Hamzehlou, M. Darroudi, J. Verdi, L. Hasanzadeh, H.-W. Kim and M. Mozafari, *Nanomedicine*, 2018, **13**, 3051–3069.
- 21 Y. Gao and X. C. Zeng, *J. Am. Chem. Soc.*, 2005, **127**, 3698–3699.
- 22 X. Dong, S. Jalife, A. Vásquez-Espinal, E. Ravell, S. Pan, J. L. Cabellos, W.-y. Liang, Z.-h. Cui and G. Merino, *Angew. Chem., Int. Ed.*, 2018, **57**, 4627–4631.
- 23 X. Ma, Y. Bai, Y. Song, Q. Li, Y. Lv, H. Zhang, H. Yu and M. Zhu, *Angew. Chem., Int. Ed.*, 2020, **59**, 17234–17238.
- 24 Y. Gao, S. Bulusu and X. C. Zeng, *ChemPhysChem*, 2006, **7**, 2275–2278.
- 25 B. Luan, T. Huynh and R. Zhou, *J. Chem. Phys.*, 2015, **142**, 234102.
- 26 H. Gao, A. Otero-de-la-Roza, J. Gu, D. A. Stone, S. M. Aouadi, E. R. Johnson and A. Martini, *ACS Appl. Mater. Interfaces*, 2015, **7**, 15422–15429.
- 27 F. G. Sen, A. Kinaci, B. Narayanan, S. K. Gray, M. J. Davis, S. K. R. S. Sankaranarayanan and M. K. Y. Chan, *J. Mater. Chem. A*, 2015, **3**, 18970–18982.
- 28 G. Kresse and J. Furthmüller, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**, 11169–11186.
- 29 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- 30 S. L. Dudarev, G. A. Botton, S. Y. Savrasov, C. J. Humphreys and A. P. Sutton, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1998, **57**, 1505–1509.
- 31 A. Singraber, T. Morawietz, J. Behler and C. Dellago, *J. Chem. Theory Comput.*, 2019, **15**, 3075–3092.
- 32 J. Behler and M. Parrinello, *Phys. Rev. Lett.*, 2007, **98**, 146401.
- 33 J. Behler, *J. Chem. Phys.*, 2011, **134**, 074106.
- 34 T. B. Blank and S. D. Brown, *J. Chemom.*, 1994, **8**, 391–407.
- 35 A. P. Thompson, H. M. Aktulga, R. Berger, D. S. Bolintineanu, W. M. Brown, P. S. Crozier, P. J. in 't Veld, A. Kohlmeyer, S. G. Moore, T. D. Nguyen, R. Shan, M. J. Stevens, J. Tranchida, C. Trott and S. J. Plimpton, *Comput. Phys. Commun.*, 2022, **271**, 108171.

