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## Pushing the boundary of the stability and band gap Pareto front by going towards high-entropy perovskites†

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Lead-free  $\text{Cs}_2\text{BX}_6$  ( $\text{B} = \text{Zr}^{4+}, \text{Sn}^{4+}, \text{Te}^{4+}, \text{Hf}^{4+}, \text{Re}^{4+}, \text{Os}^{4+}, \text{Ir}^{4+}$ , and  $\text{Pt}^{4+}$  and  $\text{X} = \text{Cl}^-, \text{Br}^-,$  and  $\text{I}^-$ ) vacancy-ordered double perovskites have gained significant attention due to their high performance in solar cell devices. Besides mitigating toxicity concerns associated with the use of lead, the presence of a formally tetravalent B-site in  $\text{Cs}_2\text{BX}_6$  has been demonstrated to improve the stability against air and moisture. Recently, experimental studies have shown that high-entropy forms of vacancy-ordered double perovskites can be synthesized and stabilized at room temperature, which opens new opportunities for designing better solar cell absorbers. In this work, we employed high throughput density functional theory (DFT) calculations using the HSE06 hybrid functional to study 546 medium-to-high-entropy vacancy-ordered double perovskites. Our results show that  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\text{X}_6$  and  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\{\text{XX}'\}_6$  perovskites can break the existing linear scaling relationships between the bandgap and formation energy observed in the pure  $\text{Cs}_2\text{BX}_6$  and  $\text{Cs}_2\text{B}\{\text{XX}'\}_6$  perovskites, which enables materials that simultaneously exhibit an optimal band gap of  $\sim 1.3$  eV for single-junction solar cells along with a low formation energy. Electronic structure analysis reveals that this can be attributed to the weak coupling between the  $\text{BX}_6$  octahedra in  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\text{X}_6$  and  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\{\text{XX}'\}_6$ . Based on these findings, we identified the analytical equations that can be used to efficiently predict the band gap and formation energy of high-entropy perovskites from their constituent pure perovskites. Our study offers simple and practical guidelines for the design and synthesis of novel high-entropy perovskites with improved photovoltaic performance.

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## Introduction

At present, lead halide perovskites are the best performing solution-processed photovoltaic materials with a power conversion efficiency record of 26.7% in single-junction perovskite solar cells and 34.6% in perovskite/Si 2T tandem solar cells in 2024.<sup>1–5</sup> However, the ultimate commercialization of these materials still faces serious practical challenges due to their toxicity from lead and their instability against moisture, air, and temperature.<sup>6–10</sup> Alternative compositional and structural derivatives of the perovskite family have consequently been explored to assess their potential in photovoltaic applications.<sup>11–16</sup> Among these novel perovskites developed over the past decade,  $\text{Cs}_2\text{BX}_6$  ( $\text{B} = \text{Zr}^{4+}, \text{Sn}^{4+}, \text{Te}^{4+}, \text{Hf}^{4+}, \text{Re}^{4+}$ ,

$\text{Os}^{4+}, \text{Ir}^{4+}$ , and  $\text{Pt}^{4+}$  and  $\text{X} = \text{Cl}^-, \text{Br}^-,$  and  $\text{I}^-$ ) vacancy-ordered double perovskites have attracted particular attention following the recent demonstrations of their high performance in solar cell devices.<sup>17–21</sup> Besides mitigating toxicity concerns associated with the use of lead, the presence of a formally tetravalent B-site in  $\text{Cs}_2\text{BX}_6$  can also improve air and moisture stability relative to the divalent B-site in conventional  $\text{CsBX}_3$  perovskites.<sup>22–24</sup>

High-entropy halide perovskites are an emerging class of materials that combine the intriguing properties of halide perovskites with the concept of high entropy.<sup>25,26</sup> Common advantages of high-entropy materials, which contain five or more different elements distributed in a single crystal structure, include entropic stabilization (at high temperature) and a tunable continuum of materials properties.<sup>25,27–29</sup> This innovative design strategy has been widely explored in metals and oxides,<sup>30–33</sup> but its application to halide perovskites is a relatively new concept and holds great untapped potential. A recent experimental study has demonstrated a viable route to synthesize and stabilize high-entropy vacancy-ordered double perovskites at room temperature such as the five-element  $\text{Cs}_2\{\text{SnTeReIrPt}\}_1\text{Cl}_6$  and six-element  $\text{Cs}_2\{\text{SnTeReOsIrPt}\}_1\text{Cl}_6$  compositions.<sup>25</sup> This opens up new opportunities for design-

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**Fig. 1** The chemical space and atomic structures of the pure and mixed vacancy-ordered double perovskites investigated in this work.

ing novel materials based on high-entropy vacancy-ordered double perovskites for photovoltaic devices beyond the state-of-the-art lead halide perovskites.

In this work, we performed high throughput density functional theory (DFT) calculations using the hybrid functional HSE06 to study over five hundred medium-to-high-entropy vacancy-ordered double perovskites. We calculated the band gap and formation energy of these perovskites and found that high-entropy perovskites exhibit emerging properties that are not present in their pure counterparts. We further analyzed their density of states (DOS) and revealed that this can be attributed to the unique electronic structure of vacancy-ordered double perovskites with weakly coupled octahedra.

Finally, we identified the analytical equations that can be used to efficiently predict the band gap and formation energy of high-entropy perovskites using the values of their constituent pure perovskites. Our work demonstrated new opportunities for applications of high-entropy halide perovskites in solar cell devices.

## Results and discussion

Vacancy-ordered double perovskites are a specialized class of perovskite materials in which specific atomic sites within the crystal structure are deliberately left vacant, leading to a unique arrangement of atoms. In conventional  $Cs_2BB'X_6$  ( $B = Cu^+, Ag^+, Au^+, K^+, In^+$ , and  $Ti^+$  and  $B' = Bi^{3+}, Sb^{3+}$ , and  $Tl^{3+}$ ) double perovskites, two different cations B and B' occupy alternating positions within the lattice. However, in  $Cs_2BX_6$  vacancy-ordered double perovskites, as shown in Fig. 1, one of the cation sites is left vacant, creating a pattern of missing atoms within the structure. In this work, we considered 8 elements (Zr, Hf, Re, Os, Ir, Pt, Sn and Te) at the B-site, and 3 elements (Cl, Br, and I) at the X-site. Besides the pure  $Cs_2BX_6$  perovskites, we also investigated three types of mixed perovskites: (1)  $Cs_2B\{XX'\}_6$  with the X-site mixed, (2)  $Cs_2\{B_1B_2B_3B_4\}X_6$  with the B-site mixed, and (3)  $Cs_2\{B_1B_2B_3B_4\}\{XX'\}_6$  with both B- and X-sites mixed. For X-site mixing, we examined the combinations of  $Cl_4Br_2$ ,  $Cl_2Br_4$ ,  $Br_4I_2$ , and  $Br_2I_4$ , considering their atomic radii and structural symmetry.<sup>25,34</sup> Previous studies have shown that the large difference in atomic radii between the  $Cl^-$  and  $I^-$  anions leads to low ionic miscibility, preventing  $Cl^-$  from substituting into the  $I^-$  lattice, particularly for relatively high concentration levels.<sup>35–37</sup> This in total yields 546 perovskite structures: (1) 24 pure  $Cs_2BX_6$  perovskites (8 elements at the B-site  $\times$  3 elements at the X-site), (2) 32  $Cs_2B\{XX'\}_6$  perovskites (8 elements at the



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B-site  $\times 4$  combinations at the X-site), (3) 210  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\text{X}_6$  perovskites (70 combinations at the B-site  $\times 3$  elements at the X-site), and (4) 280  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\{\text{XX}'\}_6$  perovskites (70 combinations at the B-site  $\times 4$  combinations at the X-site).

Fig. 2 presents the DFT-calculated formation energy and band gap for the 546 vacancy-ordered double perovskites. The computed values for the lattice constant, band gap, formation enthalpy, entropy, and Gibbs free energy at 298 K are provided in Table S1.† A linear scaling relationship between them can be observed for  $\text{Cs}_2\text{BX}_6$  and  $\text{Cs}_2\text{B}\{\text{XX}'\}_6$  perovskites: the band gap increases as the magnitude of the formation energy increases. This has been widely reported for vacancy-ordered double perovskites and can be roughly explained by molecular orbital theory.<sup>38–40</sup> Typically, perovskites with stronger chemical (covalent) bonds have more negative formation energies (*i.e.* larger magnitude) along with a wider splitting between the bonding and anti-bonding molecular orbitals (*i.e.* larger band gap). As a result, perovskites with desirable band gaps (1.1–1.5 eV)<sup>41</sup> for light absorption normally exhibit relatively poor stability. As highlighted by a grey box in Fig. 2, only very few  $\text{Cs}_2\text{BX}_6$  and  $\text{Cs}_2\text{B}\{\text{XX}'\}_6$  perovskites are in this band gap range and their formation energy are relatively high, corresponding to low stability. Table 1 lists the chemical composition, band gap, and formation energy of these perovskites. Our computational predictions of the band gaps agree very well with previous computational and experimental studies.<sup>22,40,42–44</sup> It is also

worth noting that all these perovskites are made of expensive metals (*i.e.*, Pt, Os, or Ir). However, very excitingly, we found that the B-site-mixed  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\text{X}_6$  and  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\{\text{XX}'\}_6$  medium-to-high-entropy perovskites can break the linear scaling relationship between the formation energy and band gap, allowing for significantly more candidates exhibiting an optimal band gap and high stability for solar absorbers.

Table S2† presents 99 promising vacancy-ordered double perovskites with band gaps ranging from 1.1 to 1.5 eV. It is important to highlight that, as solar-absorbing materials, these perovskites may exhibit dipole-allowed transition gaps (referred to as “allowed gaps”) that are larger than their direct band gaps.<sup>45</sup> For example, the direct band gap of  $\text{Cs}_2\text{SnI}_6$  was calculated to be 0.7 eV, while its dipole-allowed transition gap was found to be 1.1 eV. Additionally, these materials may exhibit significant exciton binding energies, which can impede charge separation and reduce open-circuit voltages in solar cells.<sup>46</sup> Therefore, in addition to the fundamental band gaps (VBM–CBM) shown in Fig. 2, it is crucial to further investigate the optical band gap and exciton binding energy of these promising candidates in future studies. Moreover, while formation energy provides an initial estimate of the stability of predicted perovskites, phonon calculations and finite-temperature molecular dynamics simulations can be further employed for a more comprehensive evaluation of their dynamic and thermodynamic stability.

We then sought to understand the origin of the breaking of the linear scaling relationship between the formation energy and band gap of the B-site-mixed  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\text{X}_6$  perovskites. As shown in Fig. 3a, we found that the formation energy of  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\text{X}_6$  can be predicted very well by the average of the formation energy of the constituent pure perovskites using the following equation:

$$E_{\text{form}}(\text{B}_1, \text{B}_2, \text{B}_3, \text{B}_4) = \frac{1}{4} \sum_{i=1}^4 E_{\text{form}}(\text{B}_i) \quad (1)$$

where  $E_{\text{form}}(\text{B}_1, \text{B}_2, \text{B}_3, \text{B}_4)$  is the formation energy of  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\text{X}_6$  and  $E_{\text{form}}(\text{B}_i)$  is the formation energy of the constituent pure perovskite, whereas for the band gap, the corresponding equation does not work:

$$E_{\text{g}}(\text{B}_1, \text{B}_2, \text{B}_3, \text{B}_4) = \frac{1}{4} \sum_{i=1}^4 E_{\text{g}}(\text{B}_i) \quad (2)$$

where  $E_{\text{g}}(\text{B}_1, \text{B}_2, \text{B}_3, \text{B}_4)$  is the band gap of  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\text{X}_6$  and  $E_{\text{g}}(\text{B}_i)$  is the band gap of the constituent pure perovskite. As illustrated in Fig. 3b, eqn (2) generally overestimates the

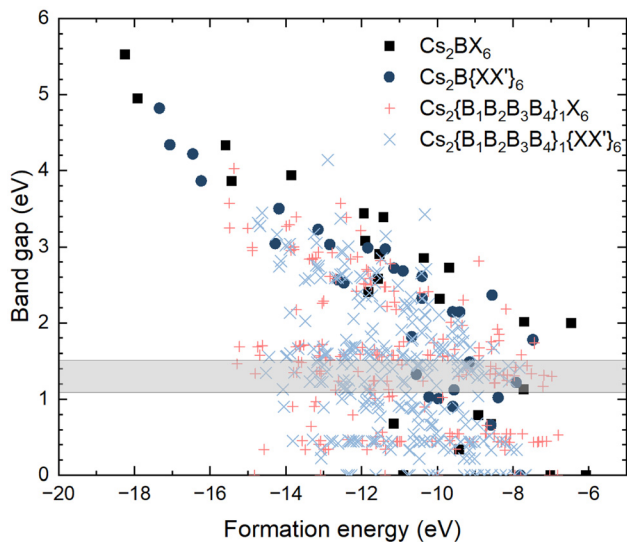


Fig. 2 DFT-calculated formation energy and band gap for 546 vacancy-ordered double perovskites including  $\text{Cs}_2\text{BX}_6$ ,  $\text{Cs}_2\text{B}\{\text{XX}'\}_6$ ,  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\text{X}_6$ , and  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\{\text{XX}'\}_6$ .

Table 1 The band gap  $E_{\text{g}}$  and formation energy  $E_{\text{f}}$  of the five  $\text{Cs}_2\text{BX}_6$  and  $\text{Cs}_2\text{B}\{\text{XX}'\}_6$  perovskites with 1.1–1.5 eV band gaps

	$\text{Cs}_2\text{PtI}_6$	$\text{Cs}_2\text{PtBr}_4\text{I}_2$	$\text{Cs}_2\text{OsCl}_2\text{Br}_4$	$\text{Cs}_2\text{OsBr}_4\text{I}_2$	$\text{Cs}_2\text{IrCl}_4\text{Br}_2$
$E_{\text{g}}$ (eV)	1.13	1.49	1.12	1.22	1.33
$E_{\text{f}}$ (eV)	−7.72	−9.15	−9.55	−7.91	−10.55





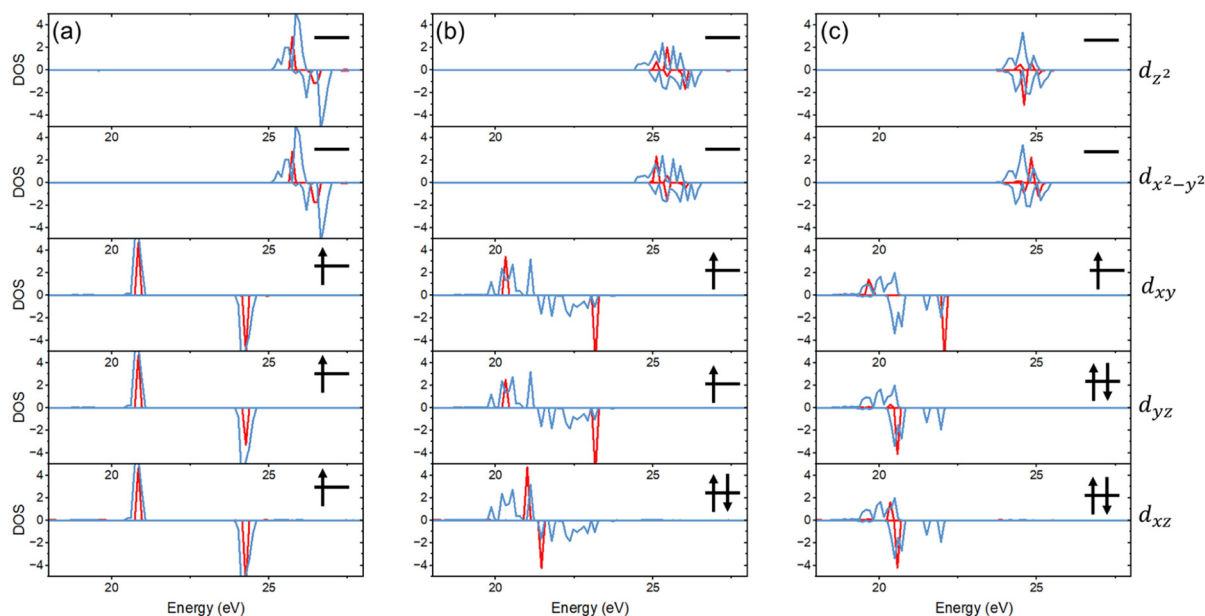
**Fig. 3** The comparison between (a) the predicted formation energy from eqn (1) and the DFT-calculated formation energy and (b) the predicted band gap from eqn (2) and the DFT-calculated band gap.

band gap of  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\text{X}_6$  with the predicted values consistently exceeding those calculated using DFT.

The dramatically different correlations with the constituent pure perovskites for the formation energy and band gap shown in Fig. 3 explain the breaking of scaling observed in Fig. 2. To gain a better understanding of the correlation of the band gap of  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\text{X}_6$  with their constituent pure perovskites, we performed a detailed analysis of their electronic structures. We found that for the diamagnetic metal ions (*i.e.*,  $\text{Hf}^{4+}$ ,  $\text{Pt}^{4+}$ ,  $\text{Sn}^{4+}$ ,  $\text{Te}^{4+}$ , and  $\text{Zr}^{4+}$ ), the projected density of states is very similar in the mixed  $\text{Cs}_2\{\text{B}_1\text{B}_2\text{B}_3\text{B}_4\}_1\text{X}_6$  and pure  $\text{Cs}_2\text{BX}_6$  perovskites. This also applies to the paramagnetic metal ion  $\text{Re}^{4+}$  with each  $t_{2g}$  orbital being singly occupied (Fig. 4a). As for the

paramagnetic metal ions  $\text{Ir}^{4+}$  and  $\text{Os}^{4+}$ , we found that the pure perovskites exhibit degenerate  $t_{2g}$  orbitals (*i.e.*,  $d_{xz}$ ,  $d_{yz}$ , and  $d_{xy}$  orbitals), which agrees with previous reports.<sup>47–49</sup> However, as shown in Fig. 4b and c, in the mixed perovskites, the breaking of symmetry leads to splitting of the  $t_{2g}$  orbitals. It is noted that while the  $t_{2g}$  orbitals are different for the mixed and pure perovskites, the energy range of the projected density of states is not significantly changed.

The highly overlapped energy range of the electronic states in the pure and B-site-mixed perovskites shown in Fig. 4 can be attributed to the relatively weak coupling between the neighboring  $\text{BX}_6$  octahedra in the vacancy-ordered double perovskites. Consequently, the conduction band minimum (CBM)



**Fig. 4** (a) Comparison of the projected density of states on Re 5d orbitals in  $\text{Cs}_2\text{ReCl}_6$  (blue) and  $\text{Cs}_2\{\text{ZrSnRePt}\}_1\text{Cl}_6$  (red). (b) Comparison of the projected density of states of Os 5d orbitals in  $\text{Cs}_2\text{OsCl}_6$  (blue) and  $\text{Cs}_2\{\text{ZrSnOsPt}\}_1\text{Cl}_6$  (red). (c) Comparison of the projected density of states on Ir 5d orbitals in  $\text{Cs}_2\text{IrCl}_6$  (blue) and  $\text{Cs}_2\{\text{ZrSnIrPt}\}_1\text{Cl}_6$  (red).





**Fig. 5** Alignment of the density of states for the B-site-mixed  $\text{Cs}_2(\text{ZrSnTeHf})_1\text{Cl}_6$  perovskite and its constituent pure perovskites:  $\text{Cs}_2\text{ZrCl}_6$ ,  $\text{Cs}_2\text{SnCl}_6$ ,  $\text{Cs}_2\text{TeCl}_6$ , and  $\text{Cs}_2\text{HfCl}_6$ .

and valence band maximum (VBM) of the pure perovskites can be used to estimate the band gap of the B-site-mixed perovskites. Fig. 5 shows the alignment of the density of states between the B-site-mixed and the constituent pure perovskites using  $\text{Cs}_2(\text{ZrSnTeHf})_1\text{Cl}_6$  as an example. As one can see, the band gap of a B-site-mixed perovskite is essentially determined by the highest VBM and the lowest CBM of the constituent pure perovskites, as expressed by the following equation:

$$E_{\text{gap}}(B_1, B_2, B_3, B_4) = \min(\text{CBM}_{B_1}, \text{CBM}_{B_2}, \text{CBM}_{B_3}, \text{CBM}_{B_4}) - \max(\text{VBM}_{B_1}, \text{VBM}_{B_2}, \text{VBM}_{B_3}, \text{VBM}_{B_4}) \quad (3)$$

where  $E_{\text{gap}}(B_1, B_2, B_3, B_4)$  is the band gap of the B-site-mixed perovskite.  $\text{CBM}_{B_i}$  refers to the CBM and  $\text{VBM}_{B_i}$  refers to the

VBM of the constituent pure perovskites, respectively ( $i = 1, 2, 3,$  and 4). We further analyzed the 99 promising vacancy-ordered double perovskites listed in Table S2.† Notably, 94 of them contain either Ir or Os, indicating that these two elements are particularly beneficial for achieving the desired band gap. This can be attributed to the unique band edges of  $\text{Cs}_2\text{IrCl}_6$  and  $\text{Cs}_2\text{OsCl}_6$ . As shown in Fig. S1,†  $\text{Cs}_2\text{IrCl}_6$  exhibits the lowest CBM, while  $\text{Cs}_2\text{OsCl}_6$  has the highest VBM among the eight constituent pure perovskites. The desired band gap of 1.1–1.5 eV can only be achieved when either Ir or Os is present.

We then applied eqn (3) to all B-site-mixed  $\text{Cs}_2\{B_1B_2B_3B_4\}_1X_6$  and  $\text{Cs}_2\{B_1B_2B_3B_4\}_1\{XX'\}_6$  perovskites, and Fig. 6a shows the comparison between the predicted band gap from eqn (3) and the DFT-calculated band gap. We found that



**Fig. 6** Comparison of the predicted and DFT-calculated band gaps for  $\text{Cs}_2\{B_1B_2B_3B_4\}_1X_6$  and  $\text{Cs}_2\{B_1B_2B_3B_4\}_1\{XX'\}_6$  perovskites. (a) Predictions using pure perovskite band edges. Red circles represent perovskites with multiple paramagnetic metal ions (e.g.,  $\text{Re}^{4+}$ ,  $\text{Os}^{4+}$ , and  $\text{Ir}^{4+}$ ). (b) Predictions using pure perovskite band edges with the red circles removed. (c) Predictions using band edges obtained from randomly chosen B-site-mixed perovskites.





## Conflicts of interest

There are no conflicts to declare.

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