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## Highly efficient circularly polarized electroluminescence based on chiral manganese(II) complexes†

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Currently reported circularly polarized luminescent devices are primarily based on rare earth and noble metal complexes or lead perovskite materials. Reports on electroluminescent devices employing eco-friendly luminescent materials are notably scarce. In this study, we strategically designed and synthesized manganese complexes featuring Binapo as the chiral ligand. The complex structure reveals a tetrahedral coordination configuration, with the *R/S* configurations exhibiting a mirror relationship. Leveraging the strong ligand field and chiral structural characteristics of Binapo, the enantiomers display red emission and exhibit significant circularly polarized luminescence with a circularly polarized luminescence asymmetric factor ( $g_{\text{lum}}$ ) of  $5.1 \times 10^{-3}$ . The circularly polarized electroluminescent performance was investigated by using a solution processing method and host–guest doping strategy. Our efforts resulted in device performance with an external quantum efficiency (EQE) exceeding 4%, and its electroluminescent asymmetric factor ( $g_{\text{EL}}$ ) reached an impressive  $-8.5 \times 10^{-3}$ . This surpasses the performance of most devices relying on platinum (Pt) and iridium (Ir) metal complexes and perovskite related materials. Our work establishes a pathway for the development of cost-effective and environmentally friendly chiral electroluminescent materials and devices.

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## Introduction

Circularly polarized electroluminescence (CPEL) has garnered increasing attention due to its efficient generation of circularly polarized light and its wide range of potential applications in areas such as three-dimensional (3D) displays, optical communication for spintronics, and optical data storage.<sup>1</sup> Progress in the field of chiral luminescent materials and efficient CPEL has been made since the pioneering work of Meijer *et al.* in 1997 when the first CPEL device was developed.<sup>2</sup> The most widely used chiral emitters include chiral fluorescent

materials,<sup>3</sup> chiral phosphorescent materials,<sup>4</sup> and chiral rare-earth complexes.<sup>5</sup> However, these materials are often plagued by issues such as low device efficiency or high costs due to limited raw material availability. Recently, circularly polarized thermally activated delayed fluorescence (CPTADF) materials have emerged as promising candidates for achieving efficient CPEL,<sup>6</sup> thanks to their ability to fully utilize excitons through reverse intersystem crossing processes. Nevertheless, the development of such chiral materials demands intricate structural design and labor-intensive synthesis routes. Chiral perovskite-based materials also show potential for efficient CPEL;<sup>7</sup> however, concerns regarding stability and lead toxicity may hinder their commercial viability. Therefore, developing cost-effective and eco-friendly chiral emitters that can achieve highly efficient CPEL is still desirable and urgent.

Eco-friendly manganese(II)-based complexes and hybrids have been extensively explored for various applications, owing to their affordability, ease of preparation, and excellent photophysical properties.<sup>8</sup> For example, Xiong *et al.* reported a series of ferroelectric materials with impressive performance by employing red-emissive Mn(II) hybrids.<sup>9</sup> Ma and Xia *et al.* demonstrated green-emissive Mn(II) hybrids capable of scintillation.<sup>10</sup> Mn(II) phosphors with thermochromic and vapochromic properties hold promise for anti-counterfeiting applications.<sup>11</sup> Furthermore, highly efficient organic light-emitting diodes have been developed using Mn(II) hybrids and

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complexes with strong emission.<sup>12</sup> The introduction of chiral organic ligands into emissive Mn(II) motifs has enabled the realization of circularly polarized luminescence.<sup>13</sup> To the best of our knowledge, no documented efforts have been made to achieve electroluminescence utilizing chiral manganese(II) emitters.

Here, we have successfully synthesized chiral manganese complexes, *R/S*-Mn(Binapo)Br<sub>2</sub>, with a four-coordinate configuration using chiral *R/S*-Binapo and MnBr<sub>2</sub>. Structural analysis reveals that the enantiomers exhibit mirror-symmetric structures. CD spectra also indicate chiroptical properties that are distinct from those of the phosphine ligand. Despite the four-coordinate configuration, we observed significant red emission, possibly due to the strong coordinating field of oxygen. Subsequently, we investigated their circularly polarized properties in the excited state and found a high *g*<sub>LLM</sub> of approximately  $5.1 \times 10^{-3}$ . Based on the excellent solubility of such materials, we attempted to fabricate circularly polarized electroluminescent (CPEL) devices for the first time using a solution process and host-guest doping strategy, achieving an external quantum efficiency (EQE) of up to 4.1%. Surprisingly, we obtained highly efficient circularly polarized electroluminescence with a *g*<sub>EL</sub> of  $-8.5 \times 10^{-3}$  along with good repeatability and stability, comparable or even superior to those of Pt and Ir complexes reported to date. Our work opens new avenues for

the cost-effective production of circularly polarized electroluminescence.

## Results and discussion

Starting with *R/S*-2,2'-bis(diphenylphosphino)-1,1'-binaphthyl (*R/S*-Binap), oxidation yielded *R/S*-2,2'-bis(diphenylphosphinoyl)-1,1'-binaphthyl (*R/S*-Binapo), which was then mixed with MnBr<sub>2</sub> in a mixed solvent of C<sub>2</sub>H<sub>5</sub>OH and CH<sub>2</sub>Cl<sub>2</sub> (Scheme S1 and Fig. S1–S4†). Thereafter, manganese complexes *R/S*-Mn(Binapo)Br<sub>2</sub> were obtained through a solvent-diffusion method at room temperature. Through the testing of the <sup>31</sup>P NMR spectrum of the chiral complex, we observed a significant shift in the phosphorus peak (Fig. S5 and S6†), indicating the coordination of the ligand's P=O group with Mn in the complex. The crystal structure is depicted in Fig. 1a, which crystallized in the space group of *P*2<sub>1</sub>2<sub>1</sub>2<sub>1</sub>. The crystal structure of the complex shows a distorted tetrahedral structure with Mn(II) ions coordinated to two oxygen atoms of the ligand (*R/S*-Binapo) and two bromine ions. The molecular configuration of the *R*-Mn(Binapo)Br<sub>2</sub> complex is similar to the *S*-Mn(Binapo)Br<sub>2</sub> complex but exhibits a mirror-symmetric configuration. The Mn–O bond lengths vary between 2.044(3) and 2.058(5) Å, Mn–Br bond lengths range from 2.468(1) to 2.4710(7) Å, and the bond angles formed by the four coordinating atoms with Mn(II)

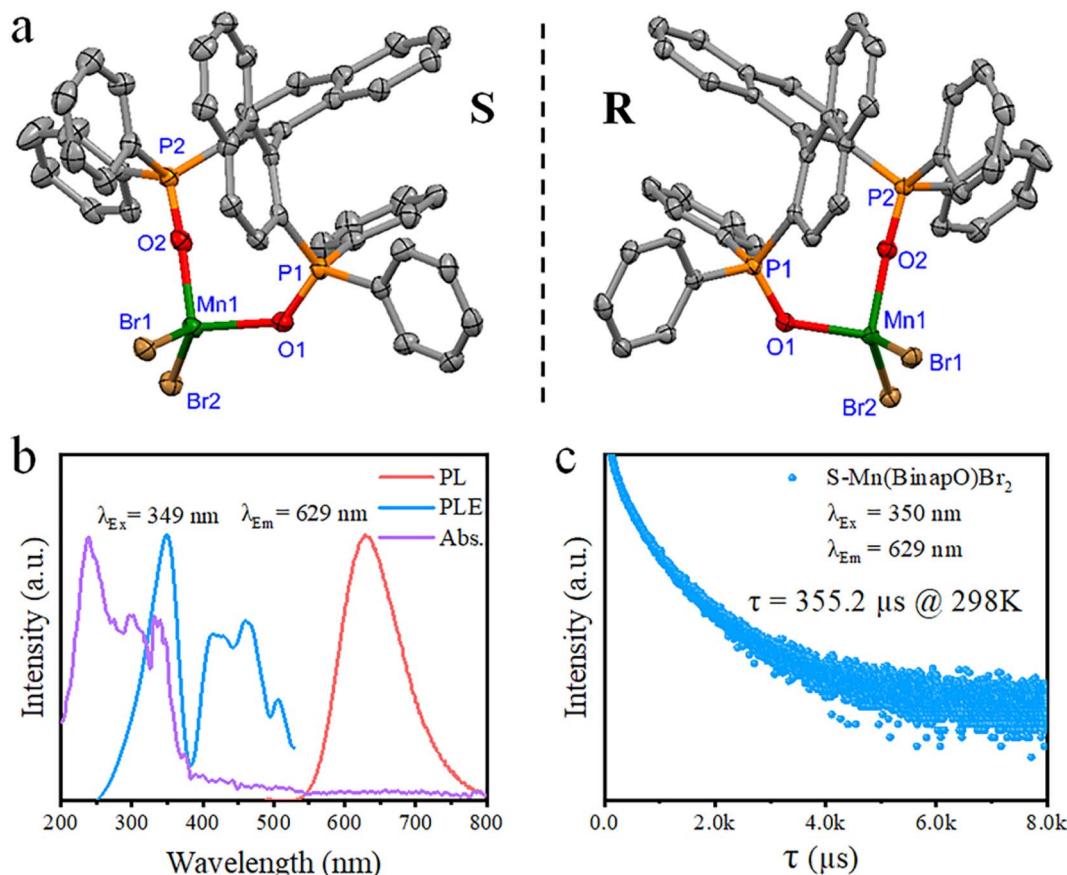


Fig. 1 (a) Visualization of the single crystal structures of the complexes *S*-Mn(Binapo)Br<sub>2</sub> (left) and *R*-Mn(Binapo)Br<sub>2</sub> (right). (b) Room temperature excitation and emission spectra and UV absorption spectra of *S*-Mn(Binapo)Br<sub>2</sub>. (c) PL decay lifetime curve of *S*-Mn(Binapo)Br<sub>2</sub>.



range from 94.9(1) to 122.78(6) $^{\circ}$  for the enantiomers. The details of crystal data are listed in Tables S1 and S2.<sup>†</sup> The powder X-ray diffraction (PXRD) pattern of the chiral complex *R/S*-Mn(Binapo)Br<sub>2</sub> is in good agreement with the simulated results from X-ray crystallography (Fig. S7<sup>†</sup>), indicating high phase purity of the prepared complexes. Elemental mapping analysis in Fig. S8<sup>†</sup> shows a uniform distribution of manganese (Mn), bromine (Br), phosphorus (P) and oxygen (O) in the *R/S*-Mn(Binapo)Br<sub>2</sub> complex. Thermal analysis reveals high thermal stability with a decomposition temperature of 450 °C (Fig. S9<sup>†</sup>).

As shown in Fig. 1b and S10,<sup>†</sup> the solid-state reflectance spectra of *R/S*-Mn(Binapo)Br<sub>2</sub> show significant absorption in the high-energy range around 200–400 nm, originating from the absorption of Binapo (Fig. S11<sup>†</sup>), attributed to  $n \rightarrow \pi^*$  and  $\pi \rightarrow \pi^*$  electronic transitions within the ligand. Weak absorption in the lower energy range from 350 to 600 nm likely arises from the forbidden d-d transitions of Mn<sup>2+</sup>. Upon excitation at around 350 nm, the enantiomers emit orange-red light with peaks at 633 nm and 629 nm. They exhibit lifetimes of 420  $\mu$ s and 355  $\mu$ s (Fig. 1c), along with photoluminescence quantum yields (PLQY) of 2.1% and 4.0% for *R*-Mn(Binapo)Br<sub>2</sub> and *S*-Mn(Binapo)Br<sub>2</sub>, respectively (Fig. 1c, S12 and S13<sup>†</sup>). The significant Stokes shift and sub-millisecond emission lifetimes indicate that the emission originates from Mn(II) d-d transitions. Significantly, the majority of tetracoordinated Mn complexes emit green

light,<sup>10a,12a,14</sup> whereas the enantiomers we synthesized emit red light. It is well known that the luminescence mechanism of manganese is primarily due to d-d transitions, and Mn–Mn coupling can also influence its luminescence. Additionally, some recent studies have reported luminescence mechanisms involving self-trapped states. We analyzed all three possible mechanisms and found that the Mn–Mn distance in our structure (8.6 Å) far exceeds the coupling distance (~6 Å), which rules out Mn–Mn coupling as a source of luminescence. Furthermore, based on the excited-state lifetime, we can also exclude self-trapped exciton emission. Then the Racah parameter (*B*) and crystal field (*D*<sub>q</sub>) were calculated to evaluate the effect of the crystal field by using the Tanabe–Sugano energy level diagram (see the details in the ESI<sup>†</sup>), which give a *B* and *D*<sub>q</sub> of 835.6 cm<sup>-1</sup> and 1059.2 cm<sup>-1</sup>, respectively, and a ratio of *D*<sub>q</sub> to *B* of 1.27, thus indicating a strong crystal field of Mn<sup>2+</sup>.<sup>15</sup> Therefore, the red emission is mainly attributable to the robust crystal field from phosphine oxide. The emission spectra recorded at different excitation wavelengths in the range of 260 to 360 nm show a maximum emission wavelength around 633 nm for *R*-Mn(Binapo)Br<sub>2</sub>, with only a single emission center as shown in Fig. S14.<sup>†</sup> The temperature-dependent emission spectra of *R*-Mn(Binapo)Br<sub>2</sub> in Fig. S15<sup>†</sup> demonstrate that as the temperature gradually decreases, the relative intensity of the emission peak increases. Importantly, there is still only one

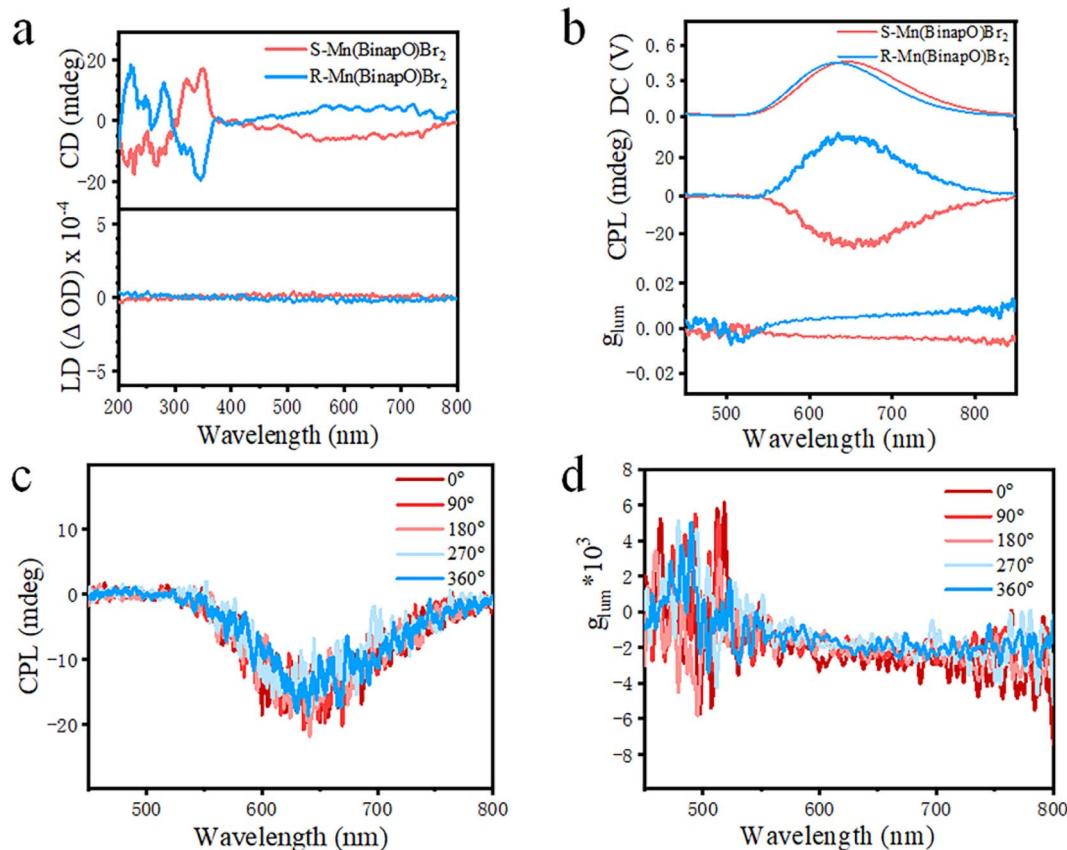


Fig. 2 (a) Circular dichroism (CD) spectrum and linear dichroism absorption (LD) spectrum of *R/S*-Mn(Binapo)Br<sub>2</sub>. (b) Circularly polarized luminescence (CPL) spectra of (*R/S*)-Mn(Binapo)Br<sub>2</sub> with DC, CPL emission and  $g_{\text{lum}}$ . (c) CPL spectra of *S*-Mn(Binapo)Br<sub>2</sub> recorded by rotating the sample from 0° to 360°. (d) Variation of  $g_{\text{lum}}$  of *S*-Mn(Binapo)Br<sub>2</sub> measured after sample rotation from 0° to 360°.



emission center, indicating that the emission solely originates from the Mn luminescent center.

Given its chiral structural features, we explored its chiroptical properties in the ground state. As shown in Fig. 2a, the enantiomers exhibit mirror-image absorption peaks in their circular dichroism (CD) spectra. Compared to the CD spectra of the ligand (Fig. S16<sup>†</sup>), the CD signal in the range of 350 to 600 nm in the complexes is attributed to Mn d-d transitions, indicating the effective transmission of chirality from the ligand to the complexes.<sup>16</sup> Moreover, we acquired linear dichroism (LD) spectra and nearly negligible signals were observed, thus eliminating the influence of LD. Thereafter, we explored its circularly polarized properties in the excited state and observed excellent CPL signals in the range of 550–800 nm under the bias of 0.4 to 0.5 V (Fig. 2b). In detail, we observed nearly symmetric CPL signals with the *R*-Mn(Binapo)Br<sub>2</sub> complex exhibiting a positive peak and the *S*-Mn(Binapo)Br<sub>2</sub> complex displaying a negative signal. The CPL signal wavelengths closely match their photoluminescence emission, confirming the chiroptical properties of these enantiomers in the excited state. To better evaluate their circular polarization performance, the photoluminescent dissymmetry factor ( $g_{\text{lum}}$ ) was introduced and calculated using eqn (1) (where  $I_{\text{L}}$  and  $I_{\text{R}}$  represent the emission intensities for left and right circular polarization, respectively). The calculation results indicate that the  $g_{\text{lum}}$  values for *R*- and *S*-Mn(Binapo)Br<sub>2</sub> complexes at their maximum emission wavelengths are  $5.1 \times 10^{-3}$  and  $-4.2 \times 10^{-3}$ , respectively.

$$g_{\text{lum}} = 2 \times (I_{\text{L}} - I_{\text{R}})/(I_{\text{L}} + I_{\text{R}}) \quad (1)$$

To eliminate the influence of linearly polarized luminescence (LPL), we measured the circularly polarized luminescence (CPL) characteristics at different angles by rotating the sample. As shown in Fig. 2c, d and S17,<sup>†</sup> during the rotating process at 0°, 90°, 180°, 270°, and 360°, the intensity and orientation of circularly polarized luminescence remained almost unchanged, indicating that the polarized emission indeed originates from CPL rather than LPL.

Notably, the complex exhibits improved luminescence efficiency in doped film states with a PLQY of 11.4% (Fig. S18<sup>†</sup>) due to the increased energy transfer efficiency from hosts and decreased phosphorescence quenching between guests. The emission of the doped film exhibited a blue shift, and as the proportion of the host material increased, the emission of the film gradually shifted further blue (Fig. S19<sup>†</sup>). Inspired by the high luminescence efficiency and good solubility, we fabricated electroluminescent devices using a solution process. The device structure is as follows: ITO/PEDOT:PSS/poly-TPD/S-Mn(Binapo)Br<sub>2</sub>/BmPyPb/LiF/Al. ITO and LiF/Al serve as the anode and cathode, while PEDOT:PSS and TmPyPb function as the hole-injection and electron-transport layers, respectively. OLED devices with the *S*-Mn(Binapo)Br<sub>2</sub> complex as the emissive layer exhibit a maximum brightness of  $31 \text{ cd m}^{-2}$ , current efficiency (CE) of  $0.045 \text{ cd A}^{-1}$ , and EQE of 0.029%, as listed in Table S3.<sup>†</sup> The poor morphology of the film as shown in Fig. S20<sup>†</sup> and imperfect exciton recombination in the emissive layer in non-doped OLEDs likely result in inferior device performance. To improve charge carrier transport and film quality, we introduced doped emissive layers into the OLEDs, selecting TAPC:2,6-DCzPPY as the doping host material. The introduction of

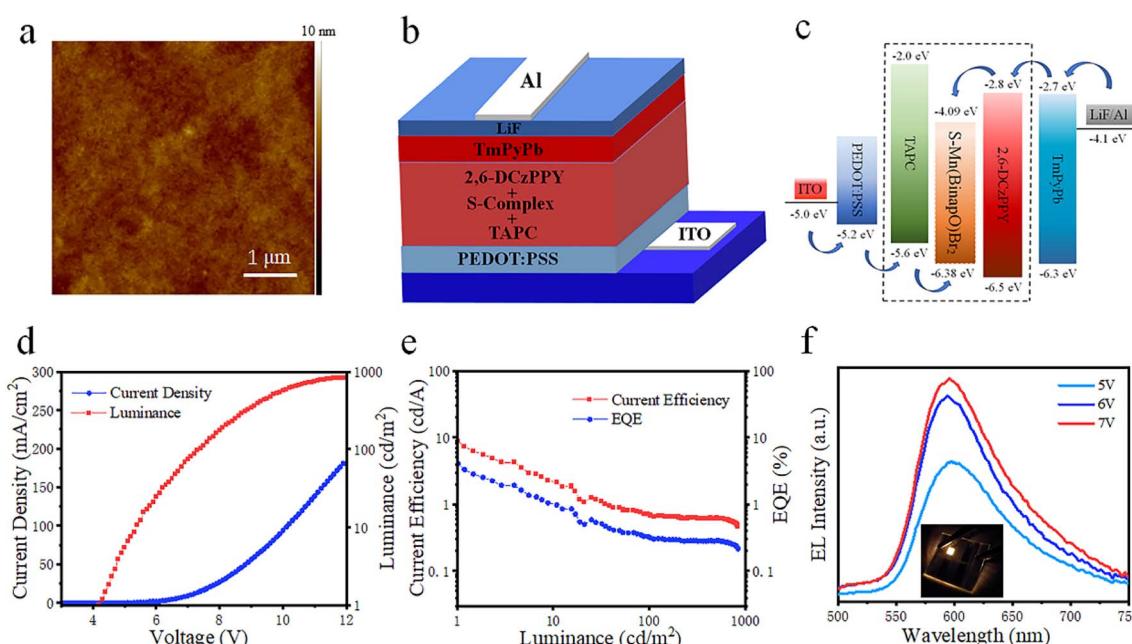


Fig. 3 (a) AFM image of the surface topography of the light-emitting layer ( $5 \times 5 \mu\text{m}$ ). (b) Schematic of the doped OLED device structure. (c) Energy level diagram of the doped OLED device. (d) Current density–voltage–luminance ( $J$ – $V$ – $L$ ) curves of the doped device. (e) Current efficiency–brightness–external quantum efficiency (CE– $L$ –EQE) curve of the dopant device. (f) EL spectra of the doped devices recorded at different voltages (with the schematic diagram of device luminescence).



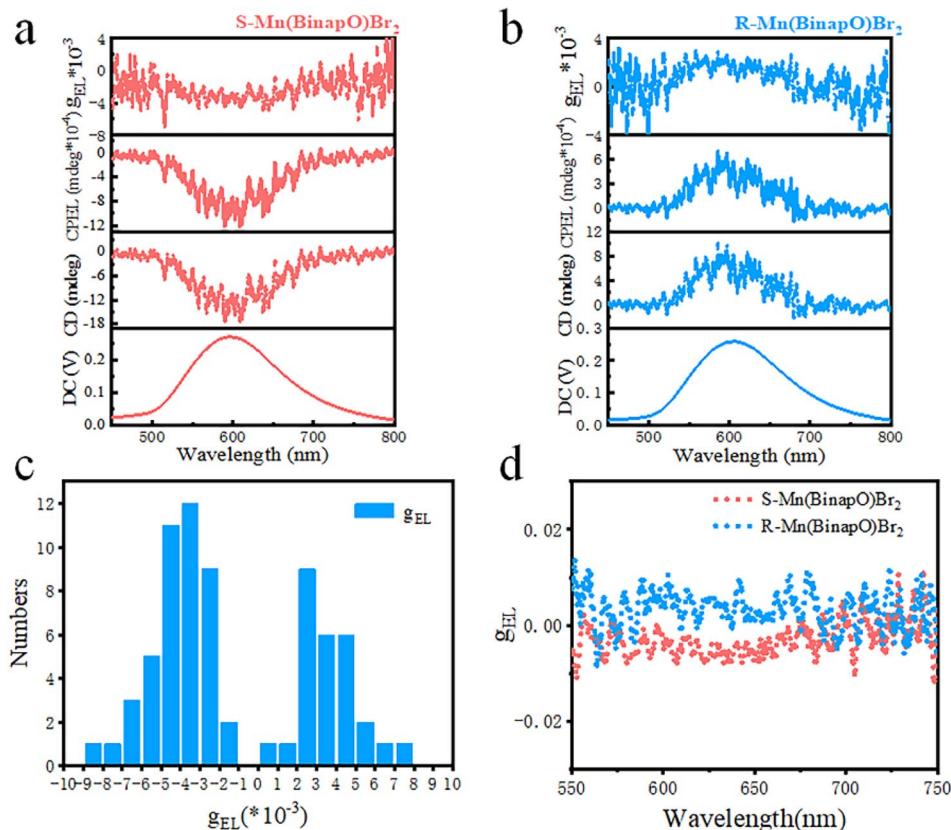


Fig. 4 Circularly polarized electroluminescence (CPEL) spectra of *S*-Mn(BinapO)Br<sub>2</sub> (a) and *R*-Mn(BinapO)Br<sub>2</sub> (b) with DC, CD, CPEL emission and  $g_{EL}$ . (c) Statistical plot of the CPEL asymmetry factor for doped OLED devices. (d) Maximum CPEL asymmetry factor curve for (R/S)-Mn(BinapO)Br<sub>2</sub>.

mixed-hole and electron transport materials into the emissive layer not only promotes charge carrier balance, but also enhances the morphology of the film, as shown in Fig. 3a and S21,† resulting in improved device performance. In comparison to non-doped OLEDs, the doped device with a structure of ITO/PEDOT:PSS(50 nm)/3% wt *S*-complex: 32.33% TAPC: 64.67% 2,6-DCzPPy (30 nm)/BmPyPb (50 nm)/LiF (1 nm)/Al (100 nm) achieves a higher brightness, CE, and EQE of 1145 cd m<sup>-2</sup>, 4.85 cd A<sup>-1</sup>, and 2.11%. However, the turn-on voltage is as high as 6.0 V. Then we replaced the electron-transporting material BmPyPb with TPBi and TmPyPb, respectively. It was found that replacing BmPyPb with TPBi resulted in a significant decrease in device performance. Although the maximum brightness increased to 2012 cd m<sup>-2</sup>, the maximum CE, power efficiency (PE), EQE, and turn-on voltage all decreased (Table S3†), mainly due to the high energy barrier between the hole injection layer and the emissive layer in the device structure. Substituting TmPyPb for BmPyPb as the electron transport material, the orange-emitting device (Fig. 3b and c) exhibited a maximum brightness of only 843 cd m<sup>-2</sup>. The device structure and energy level diagram of the optimized OLED device are shown in Fig. 3b, c and S22.† However, the highest CE, PE, EQE, and turn-on voltage in Fig. 3d and e all showed a noticeable improvement with a current efficiency of 8.79 cd A<sup>-1</sup> and an EQE of 4.09%, respectively. And the electroluminescent spectra located at around 600 nm as shown in Fig. 3f exhibit a blue shift compared

to its photoluminescence, which is common in complex electroluminescent devices.<sup>17</sup> Employing the optimized device structure, we substituted the *S*-Mn(BinapO)Br<sub>2</sub> complex with the *R*-Mn(BinapO)Br<sub>2</sub> complex and obtained a similar device performance, as shown in Fig. S23.†

Given the excellent CPL performance of the solid-state *R/S*-Mn(BinapO)Br<sub>2</sub> complexes, we conducted an in-depth study of the CPEL performance of OLED devices based on *R/S*-Mn(BinapO)Br<sub>2</sub>. The CPEL spectra of these OLED devices in Fig. 4a and b exhibited emission peaks at 597 and 603 nm, showing good agreement with the electroluminescence spectra. Similarly, we introduced the electroluminescence dissymmetry factors  $g_{EL}$  to evaluate their circular polarization characteristics. Through numerous repeated measurements as shown in Fig. 4c, we obtained an average  $g_{EL}$  value ranging from  $3 \times 10^{-3}$  to  $4 \times 10^{-3}$ , demonstrating high reliability and good repeatability for this CPEL. The highest  $g_{EL}$  values obtained in Fig. 4d for *S/R*-Mn(BinapO)Br<sub>2</sub> are  $-8.5 \times 10^{-3}$  and  $7.6 \times 10^{-3}$ , respectively. These results surpass those of most of the reported Pt and Ir complexes and perovskite materials (Table S4†).

## Conclusions

In summary, we successfully synthesized chiral manganese complexes, *R/S*-Mn(BinapO)Br<sub>2</sub>, using a chiral phosphine ligand, which emit orange emission due to the d-d transition of

Mn(II) under strong a crystal field. The introduction of the chiral ligand endows the material with chiroptical properties, resulting in a high CD and CPL signal in ground and excited states, respectively. Furthermore, we explored their circularly polarized electroluminescence by employing a host-guest doping device, which achieved an EQE of over 4% and a maximum  $g_{EL}$  of  $-8.5 \times 10^{-3}$ . All in all, we developed a CPEL device with superior circular polarization performance by employing low cost and environmentally friendly chiral Mn(II) complexes. Our work paves a promising way for constructing environmentally friendly and cost-effective OLEDs with excellent CPEL performance.

## Data availability

The data that support the findings of this study are available from the corresponding author upon reasonable request.

## Author contributions

De-Hao Kong and Hao Zeng synthesized the emitters and fabricated and characterized the OLED devices. Yue Wu and Cui-Mi Shi characterized the photophysical properties. Liang-Jin Xu provided guidance on synthesis, photophysical characterization and device fabrication. Liang-Jin Xu and De-Hao Kong wrote the manuscript. Zhong-Ning Chen supervised and directed this study.

## Conflicts of interest

There are no conflicts to declare.

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