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Insights into zero-gap CO_2 electrolysis at elevated temperatures†

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Renewable-powered CO_2 electrolysis (CO_2E) is a promising strategy to reduce greenhouse gas emissions by transforming CO_2 into valuable feedstocks. While recent studies in this field have focused on developing efficient catalyst materials or electrolyzer engineering, the operating temperature's effect has not been systematically examined for zero-gap electrolyzers. To examine the effects of operating temperature, a systematic investigation was conducted using zero-gap (MEA) Cu-based GDEs across a range from room temperature to 80 °C. Our results indicate that increasing the temperature improves CO_2 mass transport, ionic conductivity, and water management, allowing for high catalytic activity toward CO_2E. At operating temperatures greater than 50 °C, selectivity shifted substantially towards CO, with surface enhanced infrared absorption spectroscopy (SEIRAS) showing a concomitant decrease in surface CO coverage at and above this temperature. As commercial electrolyzers will operate at elevated temperatures due to ohmic heating, they may produce a significantly different product distribution than the room-temperature electrolysis prevalent in the literature. Experiments at elevated temperatures demonstrated improved results for CO_2E with industrially relevant current densities (150 mA cm^{-2}) over an extended operational period (>200 hours). Additionally, we found that the heating method strongly affects product selectivity and the electrolyzer's performance, emphasizing the need to ensure proper heating while working under these reaction systems.

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Broader context

Electrochemical CO_2 reduction is an effective method of reducing greenhouse gas emissions while creating added-value products. This process has several advantages over thermal processes, including the capability to produce multi-carbon products under ambient conditions. CO_2 electrolyzers will benefit from operating at higher temperatures due to a decrease in the activation overpotential, and improved ionic conductivity, reducing ohmic losses and overall cell potential. Through the few non-systematic studies that have investigated the effects of elevated temperature, it is known that higher temperatures are linked with an increased total current density for electrochemical CO_2 reduction at a given voltage. However, predicting the role of temperature can be challenging, as operating at higher temperatures typically results in lower CO_2 solubility and changes in diffusion coefficients and reaction rates. Additionally, there is still no general agreement on the role of temperature on product distribution, due to the lack of common testing conditions; many factors like electrocatalyst type, operating conditions, membrane stability, and cell configuration can heavily influence such trends. Herein, we conducted a systematic study of the effects of temperature on CO_2 electrolysis over Cu-based GDEs in a zero-gap cell configuration. We show that operating at elevated temperatures improved activity and selectivity towards CO_2-derived products, lowered product crossover, and enhanced stability by suppressing the hydrogen evolution reaction (HER). Operation at >50 °C constitutes a possible route for operating CO_2E cells under industrially-relevant conditions. Finally, we studied the effect of the heating methods showing its importance as a lack of proper heating technique can lead to misleading conclusions on the effect of temperature on mass transport and product distribution.

Introduction

CO_2 electrolysis (CO_2E), powered by renewable energy sources, offers a sustainable approach to converting CO_2 into chemicals and fuels.[1,2] Recently, it has been shown that CO_2E using zero-gap electrolyzers, known as membrane electrode assemblies (MEAs), can achieve industrially relevant conditions with high

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C_{2+} selectivities, reducing mass-transfer resistances and ohmic losses inherent to liquid-based CO_2 electrolyzers.[3–5] While most CO_2E research has been done under ambient conditions, commercial CO_2 electrolyzers will most likely be operated at elevated temperatures (>50 °C) due to the heat generated by overpotentials and resistive losses, especially on an industrial scale.[6] Additionally, investigating the temperature dependence on CO_2E may provide insights into its role in the mechanism of the electrochemical steps while identifying the optimal conditions for boosting electrocatalytic performance.

Although a temperature rise should improve the activity and lower the cell potential, its intrinsic effect has not been thoroughly investigated. The lack of defined trends for electrochemical CO_2 reduction (CO_2R) relates to the complex interactions between numerous factors. Temperature affects CO_2 solubility and diffusion coefficients and influences mass transport.[7–9] Furthermore, temperature changes will shift the reaction environment, equilibrium potentials, and the adsorption equilibria of intermediates, which can result in changes in electrocatalytic and homogeneous reactions (associated with the formation of the carbonate species) rates.[9,10] Temperature also improves the membrane conductivity (σ), reducing ohmic overpotentials,[11,12] and influences the water/liquid transport by modifying the electroosmotic fluxes (J_{EOD}) and evaporation rates.[13] A schematic representation of the effect of operating temperature on some reaction parameters and transport properties for CO_2R is shown in Fig. 1 and detailed in Fig. S1 and S2 (ESI†).

Despite studies incorporating temperature-dependent experiments or computational models,[14–16] there have been few systematic investigations of temperature's impact on CO_2E. The compiled data for different CO_2E studies at elevated temperatures is shown in Fig. S3 and Table S1 (ESI†). One of the first fundamental studies of temperature effects related to electrochemical CO_2 reduction (CO_2R) was conducted by Hori *et al.*[17] On Cu, they showed that CH_4 selectivity is favored at low temperatures (below ambient conditions), while CO, C_2H_4, and H_2 selectivities increase with temperature (up to 40 °C). Ahn *et al.* reported similar selectivity trends over Cu at -1.60 V *vs.* Ag/AgCl at the same operating temperatures, which they attributed to changes in CO_2 solubility and local pH caused by a higher buffer capacity at these conditions.[18] Recently, studies conducted by Vos *et al.* examined the effects of temperature, cation identity, and electrolyte on Au[9] and Cu-electrocatalysts[19] and concluded that the interaction between these factors influences the activity, product distribution (mainly towards C_1 products), and activation energy of CO_2E.

Additionally, the influence of temperature on CO_2E has been evaluated using GDEs-based electrolyzers. For example, Lowe *et al.*[7] examined how temperature impacts formate ($HCOO^-$) formation over Sn-GDEs using a flow cell electrolyzer with a catholyte layer. They found an increase in selectivity (FE_{HCOO^-} = 80%) and current density (up to 1 A cm^{-2}) by shifting from room temperature to 50 °C, while hydrogen evolution reaction (HER) dominated at 70 °C due to the interplay effects between the solubility and diffusion rates. Jeng and Jiao studied the effects of temperature on CO_2 conversion on Ag-GDE.[20] At 60 °C, they observed an increase in the CO_2 single-pass conversion due to an enhancement of the CO_2E and homogeneous reactions at elevated temperatures. Alternatively, a study by Dufek *et al.* on Ag-electrocatalyst showed a decrease in cell potential at a constant current density by increasing the temperature, with enhanced CO selectivity at 35 °C and a decrease at 70 °C due to limited CO_2 mass transport.[21] Endrodi *et al.*[15] evaluated

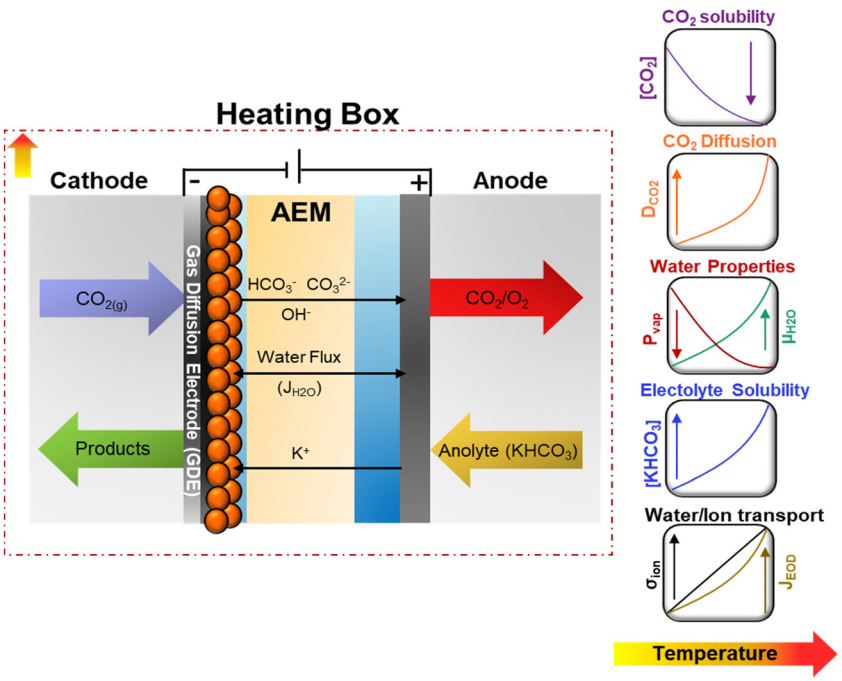

Fig. 1 Schematic representation of temperature dependency on transport and kinetics properties on CO_2E in zero-gap configuration. Equations for the presented trends for the different properties can be found in the ESI.†

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the temperature effects on PiperION anion-exchange membrane (AEM) and Garcia de Arquer *et al.*[14] evaluated the temperature effects of different ionomers on the CO_2E performance.

When investigating temperature effects on CO_2E, it is crucial to ensure uniform heating and temperature control to avoid inaccurate selectivity and water management trends. For example, a lack of uniform heating may lead to a temperature gradient across the cell and reactor components, affecting the water supply, transport, and distribution (from humidified CO_2 or the anolyte) into the system.[16] Therefore, it is essential to accurately measure and control the temperature at critical locations in the cell to evaluate its actual effect on CO_2R.

This work focused on a holistic and systematic study of temperature effects on AEM-zero gap-based CO_2E cells containing Cu-cathode GDEs. We conducted electrochemical measurements, partial pressure experiments, and *in situ* attenuated total reflectance surface enhanced infrared absorption spectra (ATR-SEIRAS) studies to evaluate how diffusion, solubility, kinetics, and mass transport vary with temperature and influence performance. Lastly, we assessed the effects of different heating methods and demonstrated the importance of proper heating control for CO_2E evaluations.

Experimental methods

Chemicals and materials

Ultra-pure water (UPW) used throughout this study was generated using a purification system (resistivity = 18.2 MΩ cm). $KHCO_3$ (Sigma-Aldrich 99.995% Trace metal basis) and $CsHCO_3$ (Sigma-Aldrich 99.995% Trace metal basis) were used for the electrolyte preparation or membrane activation before the experimental tests. $HAuCl_4$ (98.0%), NH_4Cl (99.0%), $Na_2SO_3{\cdot}Na_2S_2O_3{\cdot}5H_2O$, and NaOH were purchased from Alfa Aesar (China) Chemical Co., Ltd, while HF used to prepare the Au-films for ATR-SEIRAS was purchased from Sigma Aldrich (ACS reagent).

Electrode preparation and electrolyzer configuration

Zero-gap MEA experiments. The cathode GDEs were prepared by sputtering a 150 nm layer of 6N Cu (*i.e.* 99.9999% pure Cu) onto a commercial GDL (Sigracet 39BB from the SGL Group and acquired from fuel cell store) in a vacuum environment (10^{-6} torr) at a deposition rate of 1 Å s^{-1} under 10 sccm Ar (total pressure of 2 mtorr) and room temperature. A commercial IrO_2-coated carbon paper (from Dioxide Materials) was used as the anode-GDE. The electrochemical experiments were performed in a modified version of a commercial electrolyzer (Dioxide Materials, a grade 2 titanium serpentine flow field for the anode, and a stainless-steel flow field for the cathode side). The assembly consisted of placing a radiation-grafted AEM containing *N*-methyl-piperidinium headgroups (RG-MPIP, 55 ± 5 μm hydrated thickness in supplied Cl^- form, ion-exchange capacity IEC: 2.1 mmol g^{-1}), with a geometric area of 7.4 cm^2 between the cathode-GDE (2.25 cm^2) and anode (4 cm^2). The AEM samples were presoaked and activated in aqueous 1 M $KHCO_3$ for 24 h and then washed with UPW to remove excess ions.[22] PFA gaskets further sandwiched the MEA device, which helped prevent electrolyte leakage and short-circuiting. Finally, the system was mechanically pressed using bolts fastened with 3 N m torque to guarantee enough compression in the electrolyzer.

Electrode characterization. A scanning electron microscope (SEM) FEI Quanta 200 FEG instrument was used to characterize the electrodes and was operating at 15 kV in secondary electron mode. The catalyst's surface was further characterized by X-ray photoelectron spectroscopy (XPS) with a binding energy range of 0–1400 eV and an analyzer pass energy of 100 eV. Finally, we implemented the Probe instrument (Thermo Fisher Scientific) for data analysis, which was equipped with a hemispherical analyzer.

Electrochemical setup. For the temperature-controlled CO_2E experiments using a zero-gap cell configuration, CO_2 (AGA, purity 4.5) was supplied to the cathode flow fields with a flow rate of 20 sccm using a volumetric flow controller (Red-y from Voegtlin) after being humidified by sparging into a container filled with Millipore water. Aqueous 0.1 M electrolyte was fed from a reservoir (60 mL) to the anode and recirculated continuously using a diaphragm pump (KNF) with an approximated flow rate of 10 mL min^{-1}. Gas products in the anolyte were measured by purging Ar (at 30 sccm), allowing gaseous anolyte products to be carried along for analysis. The power source was a potentiostat (Bio-Logic VSP 300 with booster unit) operating in galvanostatic mode, and the cell potentials were reported without any *IR* corrections.

The cell and all the reaction components were placed in a heating oven with a PSU/control box interface coupled to Raspberry Pi as a PID controller. Thermocouples were placed in the electrochemical cell (cathode and anode flow fields), the humidifier, inlet, outlet streams, heating plates, and the electrolyte reservoirs. The experiments using alternative heating methods (cell not located in a heating oven) were: (1) preheating the anolyte with a heating plate coupled with a temperature controller or (2) attaching heating rods directly into the electrolyzer, controlled with a homemade PID temperature controller. Schematic representations of these heating systems can be found in ESI† (Fig. S4).

Product's quantification. Cathodic gas products were analyzed with a gas chromatograph (PerkinElmer Clarus 590) coupled with a thermal conductivity detector (TCD) and equipped with Molecular Sieve 13x and a HayeSep Q packed column using Argon (Linde 5N) as the carrier gas. The GC was also used for analyzing gases such as O_2 and CO_2 at the anode using a switching valve to measure products from both sides of the CO_2E cell during electrolysis. The partial current density (j_i) and faradaic efficiency (FE) were two primary metrics used to estimate selectivity and electrochemical performance in this study. The FE of gas products was estimated using eqn (1), while the partial current density was estimated with eqn (2).

$$\mathrm{FE} = \frac{z \cdot n \cdot F \cdot x_i}{I} \quad (1)$$

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$$j_i = \frac{I}{A} \cdot \mathrm{FE} \qquad (2)$$

where z corresponds to the number of electrons required per mol of gas product involved in the CO_2R, x_i is the mole fraction of the gas product, n is the molar flow rate in mol s^{-1}, F is the Faraday's constant (96 485 C mol^{-1}), I is the total current, and A is the electrode area. The molar flow rate was estimated by measuring the volumetric outlet flow and the ideal gas equation. The volumetric gas flow at the outlet (both anode and cathode) was measured either by using a volumetric flow meter (MesaLabs, Defender 530+) or (2) by introducing an N_2 bleeding flow (Linde, 5N) before the GC with a known concentration.

Our liquid samples were obtained from both reaction compartments. Samples were collected from the cathode after the CO_2 outlet line using a condensation unit and cold trap to measure the vapor-phase liquid products. Samples were collected from the anode from the anolyte (crossover products through the AEM). The liquid products were quantified using high-performance liquid chromatography (HPLC, Agilent 1260 Infinity II) with 5 mM H_2SO_4 as the mobile phase and 0.1 $KHCO_3$ as the reference solution. 1.5 mL aliquots of collected samples were introduced into the sampling valves. Our HPLC included a quaternary pump (up to 400 bars for eluent transport), an auto-sampler, a packed column (Aminex HPX-87X), and two detectors (refractive index detector, RID, and diode array detector, DAD).

***In situ* ATR-SEIRAS experiments.** A customized H-type electrochemical cell, separated with a bipolar membrane (Fumasep FBM from fuel cell store), was used to accommodate the Si prism and avoid cross-contamination from the counter electrode. Aqueous 0.1 M $KHCO_3$ solution was CO-saturated and preheated to the desired temperatures (room temperature to 80 °C) to be further filled in the catholyte and anolyte chambers. A mixture of CO/Ar (MKS flow controller) was fed to the cathode compartment at rates between 5–10 sccm depending on the operational temperature, varying the partial pressure to keep the CO concentration in the catholyte constant. The electrochemical cell was connected to a Bruker spectrometer equipped with an MCT detector, and a Pike Technologies VeeMAX III ATR accessory was employed for the electrochemical ATR-SEIRAS. All spectra were collected with a 4 cm^{-1} resolution and 16 scans. The reference spectrum was measured at −0.3 V *vs.* RHE in Ar-saturated $KHCO_3$. In addition, experiments were conducted using chronoamperometry at potentials between −0.3 to −0.9 V *vs.* RHE using a Bio-Logic VSP 200 potentiostat at different temperatures. The CO peak of interest in the IR spectra was detected at approximately 2050 cm^{-1}.

Results and discussions

Fundamental studies using ATR-SEIRAS

On Cu-based catalysts, CO_2E is well-known to go through a CO-intermediate to C_{2+} products.[23] CO adsorption on Cu was initially investigated using ATR-SEIRAS to determine whether it would provide fundamental insights into the temperature effect

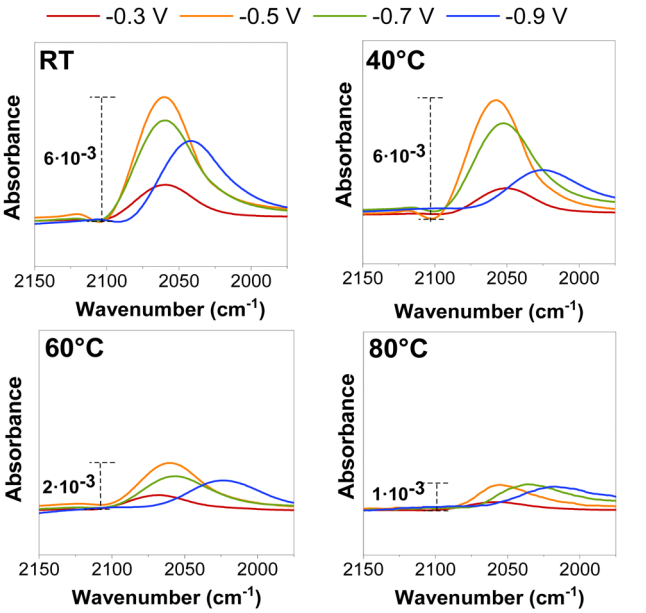

Fig. 2 ATR-SEIRAS CO adsorption peak for electrochemical CO reduction at different potentials (*vs.* RHE) and operating temperatures. The CO partial pressure for each operating temperature is detailed in Fig. S5 and S6 and estimated with eqn (S12) (ESI†).

before conducting the CO_2R in a zero-gap electrolyzer. Fig. 2 represents *in situ* ATR-SEIRA spectra in 0.1 M $KHCO_3$ for the CO signal (2050 cm^{-1}) at different cathodic potentials and temperatures (the full SEIRAS spectra for each measurement are found in Fig. S5 and S6, ESI†).

Independent of the temperature, a decrease in the CO peak position (shifting of the wavelength) and intensity was observed at more cathodic potentials (≥ -0.5 V *vs.* RHE) at potentials where there is substantial CO surface coverage, as previously reported in the literature.[24,25] These peak shifts and intensity variations may be related to chemical changes caused by either bonding changes between the adsorbed CO and the surface or to the Stark effect, which occurs when the dipoles interact with the electric field at the interface.[26] However, those effects appeared to be enhanced when the temperature was raised, as both the intensity area and stretching frequency of the CO peak decreased at all measured potentials (Fig. S7, ESI†), presumably due to a decrease in the CO coverage at the catalyst surface.

While the precise CO coverage at the surface was not quantified, we estimated the relative concentration using the intensity and area of the CO peak. Our ATR-SEIRAS spectra do not vary significantly between ambient conditions and 40 °C at less reductive potentials (< -0.5 V *vs.* RHE) as peaks do not decrease their intensity, indicating a near-saturated CO concentration at the surface. We notice a difference at very reductive potentials (> -0.9 V *vs.* RHE), but this may be on the edge of further reduction to C_{2+} products. The catalytic benefit of operating at higher temperatures may be influencing this. Interestingly, experiments at 60 °C showed a significant decrease in peak intensity due to lower CO coverages at a

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higher temperature under equilibrium conditions, assuming the same local partial pressure of CO for all the measured potentials.[13] (Variations in adsorption as a function of partial pressure are shown in Fig. S5 and S6, (ESI†)). These results give a preliminary indication that we may observe more CO–CO coupled selectivity (*i.e.*, C_{2+} products) at temperatures below 60 °C and a higher CO selectivity at temperatures above this magnitude.[19]

Temperature effects on the cathodic activity

The influence of temperature on the cathode activity was initially assessed by recording polarization curves *via* linear sweep voltammetry (LSV, 20 mV s^{-1}), as shown in Fig. 3a, and chronoamperometry measurements (Fig. S8, ESI†) using aqueous 0.1 M $KHCO_3$ as an anolyte. Polarization curves show that the temperature has a pronounced effect on the total current density, increasing approximately 3-fold at 80 °C compared to ambient temperature at the same cell potentials; this corroborates with prior observations reported in the Lowe *et al.*[7] and Vos and Koper's studies.[9]

Although the polarization curves are too short to detect products, longer-term experiments at similar temperatures and high current densities initially show a dominant selectivity towards CO_2R over H_2 evolution (HER) before the onset of any degradation or flooding mechanism during electrolysis (Fig. S9, ESI†). The exact time for flooding can be inconsistent, especially near the mass transport limit of a given electrode. (See Fig. S9 *versus* Fig. S29, ESI†). Thus, linear sweep data can be used as a proxy for CO_2 reduction activity. Moreover, we calculated the thermodynamic equilibrium potential for multiple CO_2R products, which revealed a slight negative shift at higher temperatures (−20 mV and −37 mV shifts for CO_2R to CO and C_2H_4, respectively, when rising from 20 to 80 °C). This suggests that temperature affects overpotentials linked to kinetics and mass transport more significantly than the thermodynamic potentials (Tables S2 and S3, ESI†).

The enhanced gas diffusivity and ion conductivity at high temperatures may reduce the ohmic resistances associated with the electrode and AEM. Overall, elevating the temperature lessened the cell potential (Fig. 3b) and ohmic resistances (Fig. S10, ESI†) during CO_2E. Furthermore, the cell potential decreased for all tested current densities as the temperature was increased (*e.g.*, 500 mV reduction from ambient to 80 °C at 200 mA cm^{-2}). Such voltage and ohmic resistance drops can be correlated with a higher cathode activity and ionic conductivity across the AEM (Fig. S11, ESI†).[27] However, comparing our results of the current interrupt (Fig. S10, ESI†) and recorded impedance spectra (Fig. S12, ESI†), we determined that the high-frequency resistance (HFR) decreased by less than 0.6 Ω cm^2 when the temperature was elevated, resulting in only 120 mV voltage reduction at 80 °C and 200 mA cm^{-2}. Thus, the overall cell potential change was primarily caused by improved reaction kinetics rather than changes in ionic conductivity.

Temperature effects on selectivity and product distribution

We conducted detailed chronopotentiometry (Fig. 4 and 5) and chronoamperometry (Fig. S13, ESI†) measurements to explore how temperature affects product selectivity in zero-gap configuration. Fig. S14 (ESI†) shows pre- and post-test SEM images for both room temperature and 60 °C tests from Fig. 5, whereas Fig. S15 (ESI†) shows the corresponding XPS analysis of these samples. Our experiments showed that at operating temperatures below 50 °C, there were no notable changes in the faradaic efficiencies for CO and C_{2+} products compared to those at ambient conditions (*e.g.*, 15 ± 2% and 43 ± 3% at room temperature at 200 mA cm^{-2}, for CO and C_{2+} respectively compared to 17 ± 3% and 47 ± 5% at 40 °C). However, when the temperature was raised to 50 °C, there was a drastic increase in CO selectivity with faradaic efficiencies above 40% for all the measured current densities. These results are consistent with our observations from the ATR-SEIRAS CO

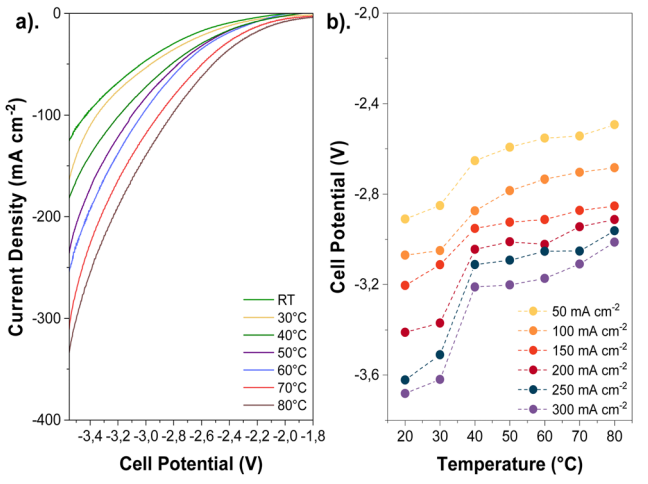

Fig. 3 Temperature effects on the cathode activity through changes in total current density and cell potential at different temperatures: (a) LSV measurements at different cell potentials and (b). Cell potential values as a function of total current density. Experimental conditions: Cu-GDE (cathode), MPIP-AEM, IrO_2 (anode), CO_2 feeding rate: 20 sccm, 0.1 M $KHCO_3$ as anolyte.

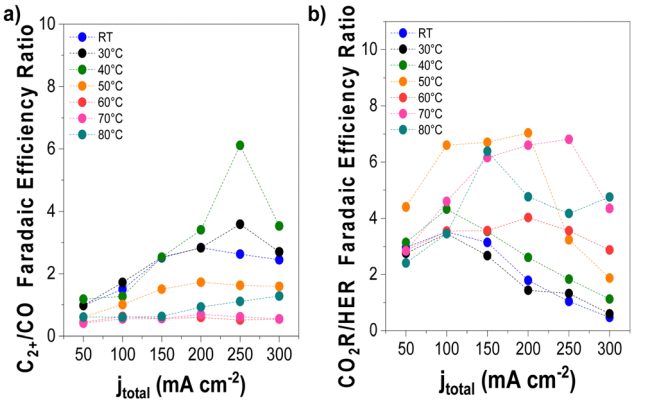

Fig. 4 Selectivity trends as a function of operating temperature and applied current density. (a) C_{2+}/CO and (b) CO_2-derived products/H_2 FE ratio. Experimental conditions: Cu-GDE (cathode), MPIP-AEM, IrO_2 (anode), CO_2 feeding rate 20 sccm, 0.1 M $KHCO_3$ as anolyte. Results are paired with those presented in Fig. 5.

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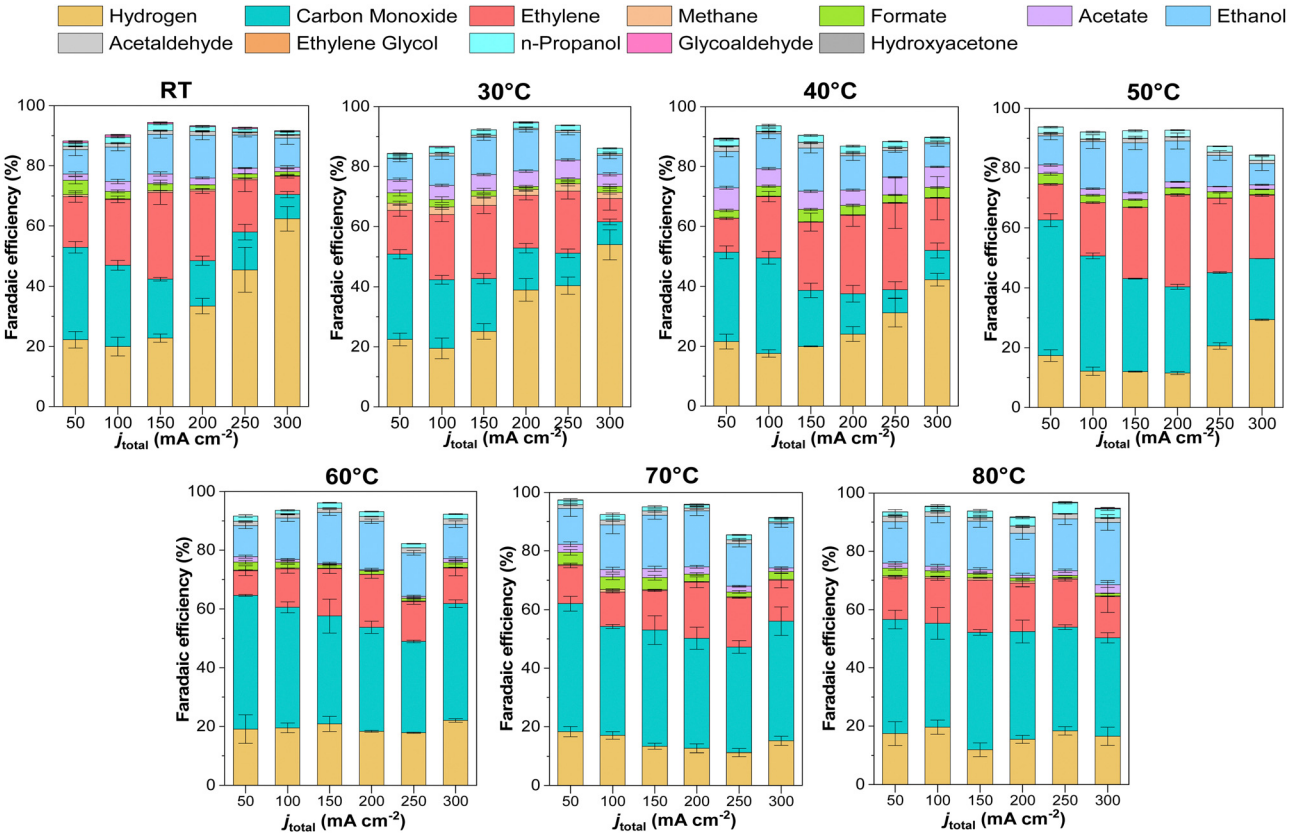

Fig. 5 Temperature effects on the product distribution for CO_2R over Cu-based electrocatalysts using zero-gap configuration at different temperatures and current densities. Experimental conditions: Cu-GDE (Cathode), MPIP-AEM, IrO_2 (Anode), CO_2 feeding rate 20 sccm, 0.1 M $KHCO_3$ as anolyte. Error bars represent the standard error of the independent measurements at each different reaction condition.

desorption studies at elevated temperatures (Fig. 2). The shift in selectivity is believed to occur as CO more rapidly desorbs from the electrode surface at elevated temperatures, leading to lower CO coverages rather than forming C_{2+} products.

The high CO selectivity at increasing temperatures could be resolved by designing a system that produces a high local CO pressure. However, the catalyst-adsorbate binding free energy scales linearly with temperature and logarithmically with pressure, which could lead to the need for excessively high CO partial pressures. As CO can form carbonyls with Ni or Fe from steel and the reaction equilibrium scales with CO pressure by a factor of 4 and 5, respectively (since it produces $Ni(CO)_4$ or $Fe(CO)_5$), this could make practical issues complicated. Alternatively, a catalyst with a slightly higher CO binding strength, such as Ni or Pd, may be more viable. However, pure Ni would struggle due to $Ni(CO)_4$, and the concomitant evaporation of the catalyst and Pd may have H-intercalation issues.[28] Although there are ways to increase CO selectivity at higher temperatures, it is uncertain if the added complexities are worthwhile.

To analyze the catalysis at the cathode in MEA devices further, we used a 5-electrode configuration pioneered by the Jiao group.[29] Using this approach, we measured potentials at 100 mA cm^{-2} and found that the cathode potential drops from −1.6 V *vs.* SHE to −1.1 V *vs.* SHE as the temperature increases from 30–80 °C (Fig. S16, ESI†). With the complexity of this method and inherent imprecisions, it is best to take these values as qualitative rather than quantitative. There was a slight increase in potential between 70–80 °C, which could be an indicator of the onset of gas diffusion layer or membrane degradation and concomitant contamination of catalysts.

Fig. 4 illustrates the selectivity trends as a function of the operating temperature in relation to C_{2+}/CO and CO_2R products/HER ratios. This allows us to see that the overall trend entails that the C_{2+}/CO ratio continues to increase as a function of total current density. Chronopotentiometry experiments are advantageous because the fixed current is accompanied by constant mass transfer and local pH despite the varying potential during electrolysis (simple Tafel analysis has shown that operating at high current densities led to higher cathodic overpotentials).[30] It is well known that the selectivity towards CO dominates over the C_{2+} products at low overpotentials; however, at higher overpotentials, the potential is reductive enough to bind CO strongly to the surface, *versus* the time needed for C–C coupling. This explains why the C_{2+}/CO ratio increases with the current density. Chronoamperometry tests also showed such trends, exhibiting a higher C_{2+}/CO ratio at larger cell potentials (Fig. S17, ESI†). The abrupt shift at 300 mA cm^{-2} at lower temperatures is attributed to the electrode flooding, thus significantly enhancing the selectivity towards HER.

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Temperature also impacts the C_{2+}/CO ratio, which generally increases to 40 °C at a constant current density before decreasing above this temperature. A shrinking C_{2+}/CO ratio at temperatures above 40 °C can easily be ascribed to the aforementioned CO desorption trends shown in Fig. 2. However, at temperatures at and below 40 °C, Fig. 2 implies a relatively constant, near-saturated CO coverage, thus the increase in the C_{2+}/CO ratio cannot be solely attributed to this phenomenon.[23] As the C–C coupling is an electrocatalytic reaction (though not a faradaic reaction), the temperature should influence its rate. Although we cannot definitively say which factor has a more significant impact on the C_{2+}/CO ratio under 40 °C, based on these results, it does appear that high CO coverage/binding strength and CO diffusion plays a role.[19]

In terms of selectivity towards specific CO_2-derived products, interestingly, as the CO selectivity increases, this seems to disproportionately decrease the ethylene selectivity compared to ethanol, possibly indicating a parameter to vary the ethylene/ethanol ratio (Fig. S17–S19, ESI†). A decrease in ethylene is observed at high temperatures, where *CO coverage is low, suggesting that *CO coverage may influence ethylene selectivity.[31] Additionally, CH_4 (Fig. S17, ESI†) was observed to be entirely suppressed at higher temperatures, in agreement with previous studies and DFT calculations, attributed to the adverse effect of temperature on proton shuttling and the stabilization of its intermediates.[10,18]

Regarding the ethanol selectivity, no major changes were observed by increasing the operation temperature (Fig. 5 and Fig. S18, ESI†). However, as expected, an increase in temperature resulted in increased evaporation of ethanol at the cathode (Fig. S18, ESI†). In general, most liquid products in our experiments were detected at the anode, with only 20–30% of products being detected by condensation in the wash stream after the cathode outlet. As the temperature increased, volatile products were transported through the GDE due to enhanced evaporation. However, at 80 °C there was still 35–50% of the total ethanol produced being extracted from the anode compartment.

It should be noted that diffusion will also increase as a function of temperature. Literature has shown that ethanol's diffusion coefficient in water will increase by a factor of $\sim$3 going from 25 °C to 85 °C. However, at the same temperature range, the vapor pressure will increase by a factor of 16, entailing that improved evaporation at higher temperatures will be the dominant factor in whether ethanol stays on the cathode or migrates to the anode.[32] Hence, a temperature rise may represent a strategy to concentrate ethanol at the cathode compartment, thereby reducing the separation costs of CO_2 derived products.[5,33] Negatively charged compounds do not experience a change in their crossover rate due to electromigration across the AEM and lack of a vapor pressure. Traces of formate and acetate were found in the cathode wash, suggesting that these non-volatile products had been transferred *via* microbubbles during gas stripping.[34]

H_2 selectivity shows an exciting trend with temperature. We observed that a low-temperature regime ($<$ 40 °C) has a marked increase in the H_2 faradaic efficiency; however, when increasing the temperature, the selectivity unexpectedly drops. Excessive selectivity towards HER is usually a result of cathode flooding, often accompanied by mass transfer issues at higher current densities. In our system, however, HER selectivity decreased at temperatures beyond 50 °C, indicating that CO_2 mass transport limitations are not the primary determinant of HER selectivity under these operating conditions, but rather the water management. Water management and flooding issues are complex in CO_2 electrolysis devices and depend on multiple factors, including the membrane. In MEA-cell configurations, AEMs do not directly influence catalytic activity. Still, they can modify the local reaction environment and water transport and indirectly influence product selectivity, especially when comparing CO_2-reduced products *versus* H_2.[35] A temperature rise will increase the electroosmotic drag coefficient due to greater ion mobility and diffusion-driven transport processes across the membrane. Further, the membrane's water content is intrinsically influenced by temperature, which may alter AEM hydration, water uptake, and thus the overall water flux in the system.[30,31]

Using an AEM entails electroosmotic drag (EOD) of ions to the anion ($9H_2O/CO_3^{2-}$, $1.5H_2O/OH^-$),[36] which provides a mechanism beyond evaporation for water to be removed from the cathode. To our knowledge, the temperature dependency of EOD in AEMs has not been analyzed. In cation exchange membranes, which should follow a similar trend, temperature effects are unclear, with reports noting that EOD is either temperature-independent[37] or increases electroosmotic drag slightly.[38] Furthermore, it should be noted that even if the cathode is 100% humidified (even locally near the catalyst/membrane interface), the chemical potential difference between the cathode and membrane is negligible, thus not providing sufficient driving force (water flux) to flood the cathode-GDE.[39,40]

Even though the water concentration at the interface of the catalyst-membrane remains stable, regardless of the relative humidity at the inlet feed, there is still a small hydrostatic pressure from the anodic water (5 mbar), which is typically negligible. However, previous works have shown salt deposition can occur deeper in the GDE layer which restricts gas flow.[41,42] If our catalysts are consuming CO_2 into liquid products, C_{2+} gas products, and carbonates, this will decrease the local CO_2 pressure at the cathode. In this case, the hydrostatic pressure difference may become substantial dragging water to the cathode and flooding it. As we are incapable of measuring local pressure in the plane of the catalyst, it is hard to determine if this is a substantial effect or negligible.

Additionally, from a pure reaction stoichiometry standpoint operating at elevated temperatures with fully humidified cathodes inlet streams will mean higher absolute water contents in the CO_2 supply gas (*e.g.*, at 100% relative humidity, H_2O content is 3 mol% and 43 mol% at 25 and 80 °C, respectively).[43] As a result, both water and CO_2 concentrations in humidified gases are affected by temperature. Our exploratory experiments showed negligible differences when the relative humidity (RH)

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was varied, except for high temperatures and low RHs, where we observed a slight increase in HER selectivity (Fig. S20, ESI†), which supports some of the selectivity trends reported in the literature.[40]

Under high temperatures and low RH conditions, degradations in AEM chemistry are more visible. Higher temperatures may reduce membrane overpotential, but this also facilitates performance degradation.[44] To demonstrate this we operated at 90 °C (Fig. S21, ESI†) for 1 hour in fully humidified gas. In these tests, we used a porous silver membrane catalyst/gas diffusion layers (Sterlitech Inc., purity 99.97%)[22] as this is an excellent benchmark catalyst and eliminates any issues related to a carbon-based gas diffusion layer degradation. This allowed us to see that while commercial membranes Sustainion and PiperION were unstable, the RG-MPIP did show better stability (though still with some issues- see Fig. S21, ESI†). However, the RG-MPIP AEM did show substantial stability issues when operating at low RH levels (Fig. S22, ESI†) even at a slightly reduced temperature (80 °C instead of 90 °C). Thus, any H_2 selectivity issues at low RH may be strongly influenced by membrane degradation and, therefore, unreliable. At least partially because of this, it was found that fully humidified CO_2 inlets had the optimal performance at mitigating HER and ensuring more stable operation.

Another factor that could affect H_2 selectivity is the salt solubility in the electrolyte. A well-known fact is that cathodically produced OH^- forms carbonates, resulting in salt precipitation of carbonates such as $KHCO_3$, in our case.[42] Salt deposition blocks the flow field and GDE, which restricts CO_2 mass transfer, thus favouring hydrogen selectivity.[41] As salt solubility increases with temperature, the higher operating CO_2 electrolysis should have fewer issues with salt deposition and concomitantly less HER.[42] Reducing the salt concentration from 100 mM to 50 mM slightly diminished the H_2 selectivity (Fig. S23, ESI†). However, further decreasing this to 10 mM increased the HER selectivity, though this lower electrolyte boosted the voltage significantly (−3.4 with 50 mM *vs.* −3.8 V at 10 mM). One potential reason could be that the salt concentration may not provide a substantial enough electric field in the Helmholtz plane to allow sufficient CO_2 reduction. Another approach to mitigate salt deposition is using a more soluble salt such as Cs.[42] Switching to $CsHCO_3$ as an electrolyte showed minimal changes in H_2 selectivity (from 21% to 17% at 0.1 M $CsHCO_3$, Fig. S23, ESI†).

Temperature effects on CO_2 utilization and mass transport

Temperature directly impacts CO_2 mass transport since the CO_2 solubility in aqueous solutions decreases at higher temperatures. Therefore, we varied the CO_2 partial pressure (p_{CO_2}) using Ar at different temperatures. We adjusted the CO_2/Ar ratio for all experiments at the measured temperatures to ensure the same local CO_2 concentration during the experiments based on Henry's law (eqn (S.1)–(S.3) in ESI†). 13.5 mM was used as this coincided with the CO_2 concentration in an aqueous solution at 1 bar and 80 °C, beyond which higher concentrations would entail using a pressurized reactor. The results of the variation in partial pressure tests are shown in Fig. 6.

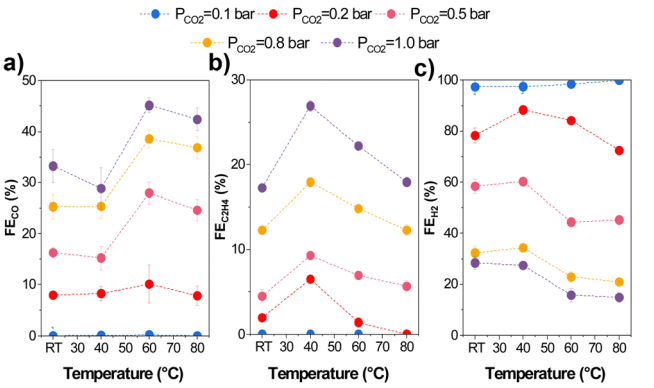

Fig. 6 Product distribution and faradaic efficiencies of gas products at different operating temperatures and partial pressures of CO_2 (diluted with Ar): (a) CO, (b) C_2H_4, and (c) H_2. Experiments were conducted at 200 mA cm^{-2}. Error bars represent the standard error of three independent measurements at each different condition.

We found increased selectivity towards carbon-derived products and lower HER at p_{CO_2} and temperatures above 0.5 bar and 60 °C. Specifically, gas product selectivity (both CO and C_2H_4) decreased with a reduction in p_{CO_2} (*e.g.*, $FE_{C_2H_4}$ = 26% at p_{CO_2} = 1 bar *vs.* 4% at p_{CO_2} = 0.2 bar), as previously reported in the literature.[45] The only significant change when varying partial pressures is a change in the CO_2R/HER ratio, which increases with an increasing p_{CO_2}. This can be explained by CO_2 being mass transport limited at low partial pressures, enhancing H_2 selectivity. Interestingly, while the C_{2+}/CO ratio varied quite substantially with temperature transitioning between 40–50 °C, the partial pressure had minimal effect on the C_{2+}/CO ratio as observed.

In addition, we investigated the effect of temperature on electrode flooding by investigating changes in electroactive surface area. In most cases, this can be accomplished by measuring double-layer capacitance (DLC). Still, since the dielectric constant (κ) varies with temperature (for water, it decreases 22% when the temperature increases from 25 °C to 80 °C),[46] we must also consider this. Thus, we measured the (DLC/κ) ratio after electrolysis at each current density and temperature (Fig. 7 and Fig. S24, ESI†). Since this work used a two-electrode cell, capacitance depends on the anode and cathode. Due to the fact that the gas diffusion electrode anode is immersed in an electrolyte that does not change with temperature or current density, we can qualitatively attribute the change in capacitance to the cathode.

We observed variations in the DLC/κ as a function of operating temperature and experimental time. Before electrolysis, DLC/κ measurements at OCV revealed steady DLC/κ and no apparent changes at constant current density (200 mA cm^{-2}). After 100 minutes of operation at lower temperatures, the DLC/κ increased, directly related to the spike in the HER. Generally, electrolyte penetration into the GDE increases gradually, causing progressive electrowetting at the electrode and influencing CO_2 diffusion at the electrode.[42,47] This is most evident at low temperatures. Higher operating temperatures resulted in a lower

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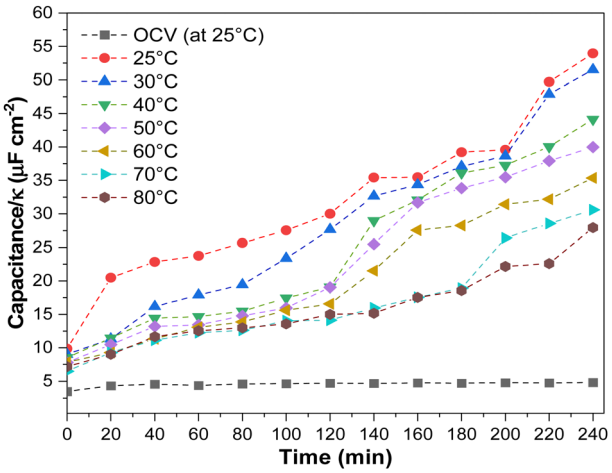

Fig. 7 Tracking the electrode flooding by DLC measurements as a function of the operating temperature at 200 mA cm^{-2}. OCV was conducted at room temperature. Experimental conditions: Cu-GDE (Cathode), MPIP-AEM, IrO_2 (Anode), CO_2 flow rate: 20 sccm, 0.1 M $KHCO_3$ as anolyte.

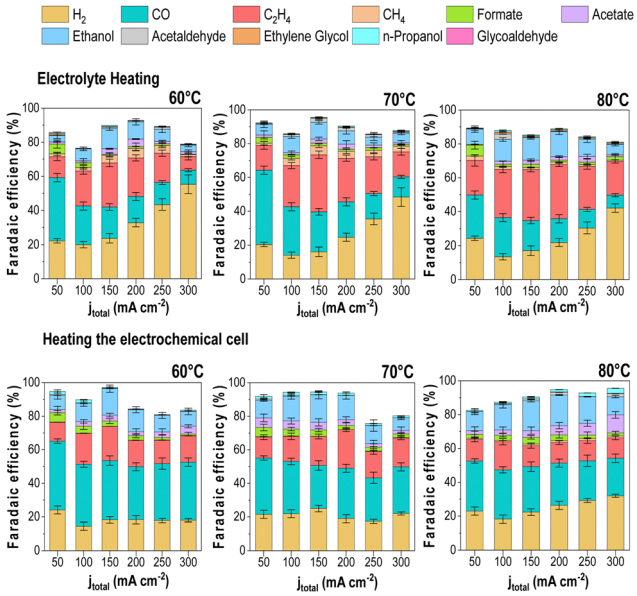

Fig. 8 Effect of the cell heating in the product distribution using two different methods at different operating temperatures experimental conditions: Cu-GDE (Cathode), MPIP-AEM, IrO_2 (Anode), CO_2 flow rate: 20 sccm, 0.1 M $KHCO_3$ as anolyte. Error bars represent the standard error of the independent measurements at each different condition.

DLC/κ increment, which may have resulted in lower GDE electrowetting (Table S4, ESI†).[47,48]

Effect of the heating method on the CO_2E performance

The lack of a standardized protocol for CO_2E electrolyzer heat control makes comparing studies at high temperatures challenging. Therefore, we decided to analyze the impact of the cell heating methods to see how this impacted our results. In addition to having the cell in a temperature-controlled heating box, we also implemented two additional heating methods for heating the reaction. The first method only heated the anolyte reservoir (60 mL) to a given temperature with all other elements of the reactor system kept at room temperature. This was to test if the anolyte (a liquid with a high heat capacity) would be enough to maintain the electrolyzer at a uniform temperature. The second method was using heating rods that were directly connected to both the cathode and anode flow fields with a thermocouple inserted into the reactor to allow the reactor to operate at a given temperature. In all methods, we waited until the temperature was equilibrated before starting the electrolysis experiments to guarantee the desired operating conditions. The results of these measurements are shown in Fig. 8. The experiments were only performed at >60 °C as variations due to different heating methods would be most apparent at these higher temperatures.

The results showed a few notable differences in the product distribution between the three heating methods. While only slight differences can be seen between the direct cell heating method and the use of the heating box (*i.e.*, Fig. 5 results), significant variations were found when heating the electrolyte only, mainly regarding increased H_2 selectivity. When we measured the cell temperature during these experiments, we found that heating the anolyte to 60 °C only led to a cell temperature of 28 °C, while a 70 °C anolyte led to a cell temperature of 32 °C, and 80 °C anolyte led to 35 °C.

When the cell temperatures are compared to Fig. 5, the results align more consistently with the observed trends. Such discrepancies between the electrolyte and the cell temperature's measured temperature might explain the variation in the overall product distribution trend for the three heating scenarios, thus demonstrating the importance of ensuring accurate cell operating temperature. Variations in faradaic efficiency between the "cell heating" method and our heating box may be due to differences in the CO_2-humidified cell and the anolyte feed. These parameters were not heated under the cell heating scenario, which likely caused temperature gradients that impacted water management and selectivity.

We must also consider that CO_2 electrolyzers generate heat from efficiency losses. Corral *et al.* described that temperature changes inside the electrolyzer alter the CO_2 concentration at the gas–liquid interface, leading to CO_2 mass transport limitations.[49] By measuring the temperature near the cathode surface, they observed a temperature increase of 7 °C under 300 mA cm^{-2} (E_{cell} = −8.7 V) and 24 °C at 500 mA cm^{-2} (E_{cell} = −14 V) for CO_2R and attributed these changes to the released heat by exothermic reactions at the surface. Considering the possible effects of resistive heating on our performance, we used a customized MEA cell that allows the placement of a thermocouple close to the GDE for more precise temperature measurement in the cell. Our room temperature experiments at 300 mA cm^{-2} had a cell potential of −3.58 V. The thermoneutral potential is approximately 1.4 V at this current density (determined *via* averaging products and their corresponding selectivity). As the electrode is 2.25 cm^2, operating at 300 mA cm^{-2} resulted in ohmic heat production of 1.5 W. This, in turn, led to a temperature rise of 3 °C at the cathode

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(Fig. S25 shows temperature rise for current densities between 50 to 300 mA cm^{-2}, ESI†). While this temperature rise is reactor-dependent, the slight change in our reactor entails that any temperature changes by resistive heating will not be significant in the system but may be significant when scaling these electrolyzers up to industrial levels.

Long-term stability of CO_2E at elevated temperatures

We performed long-term experiments (50 h) at high temperatures to track product distribution and stability (using the RG-MPIP AEM). Fig. 9 shows cell potential and product distribution of electrolysis at 60 °C (using our heating box) and 150 mA cm^{-2}. The temperature of the anode, cathode, humidifier, and electrolyte were all measured are are shown in Fig. S27 (ESI†), though all temperatures are very stable at 60 °C. The FE of gas products was calculated using the average of the simultaneous injections in the GC, and the liquid products were measured by periodically collecting anolyte and cathode samples. To avoid possible evaporation of the anolyte, we used a more extensive reservoir (400 mL *vs.* 60 mL in all other experiments) and maintained continuous operation to maximize GDE durability.

The experiment stayed relatively constant in operating potential with an initial potential of −2.8 V and a final potential of −2.9 V. Initially, CO was the primary product (FE = 39%), but this slowly decreased over the 50 h test (FE = 25%). The selectivity towards ethylene fluctuated between 10–20%, possibly with slightly lower selectivity after 50 h. Initially, the FE for ethanol was relatively high with FE = 20%, although this decayed quickly to a stable plateau of FE = 10%. From the knowledge gained from earlier in this work (Fig. S18, ESI†), we know that under the CO_2E conditions of Fig. 9, *ca.* 50% of the ethanol will evaporate off the cathode and be collected in the cold trap, and 50% will crossover to the anode. The longer the ethanol stays in the anode, the greater its chance of oxidizing. While this is unlikely to be the only cause for the measured ethanol selectivity to drop, it is potentially one significant factor. The acetate selectivity was lower at the beginning of the experiment, matching the trend presented in Fig. 3, but it was slightly higher after 24 h. This may be related to ethanol oxidation towards acetate at the anode, as complete oxidation to CO_2 is not that facile.[34] Furthermore, H_2 continually increased from 15% to 38% at the end of the experiment, showing that flooding issues were slowly worsening.

Using an Ag membrane as a catalyst/gas diffusion layer allowed us to analyse whether the drop in performance could be due to membrane issues; however Fig. S26 (ESI†) shows constant selectivity over 50 hours, suggesting the Fig. 9 changes are probably not membrane related. Another long-term issue that needs to be considered is the anode. In this work, we used IrO_2 deposited on carbon paper, and while studies in the literature show stable operations for 3000 hours,[50] thermodynamically, carbon should oxidize quite readily. Thus, for more robust stability, a mixed metal oxide (*e.g.*, Ti/IrO_2) that is the standard for water electrolysis may be a better route moving forward.

To probe the stability at high temperatures a little deeper, we did durability studies at 80 °C (see Fig. S28, ESI†), and saw substantial degradation in selectivity even using a humidified cathode over a 50-hour test. We also tested at 400 mA cm^{-2} at 60 °C, 70 °C, 80 °C and 90 °C for 4 hours while measuring gas products (see Fig. S29, ESI†). The higher current densities are derived from higher voltages, which typically favor C_{2+} products over CO. However, similar to Fig. 5, an increased potential/current density does not particulary lead to higher C_{2+} products and as the temperature increased CO selectivity increases.

As temperatures above 60 °C seem troublesome for many reasons, we reverted back to 60 °C and looked how other parameters could affect performance stability. As mentioned earlier, salt deposition can be a notable issue, contributing to performance degradation. While operating at higher temperatures increases the solubility of salts used for the electrolyte, this can be increased even more by switching from $KHCO_3$ to

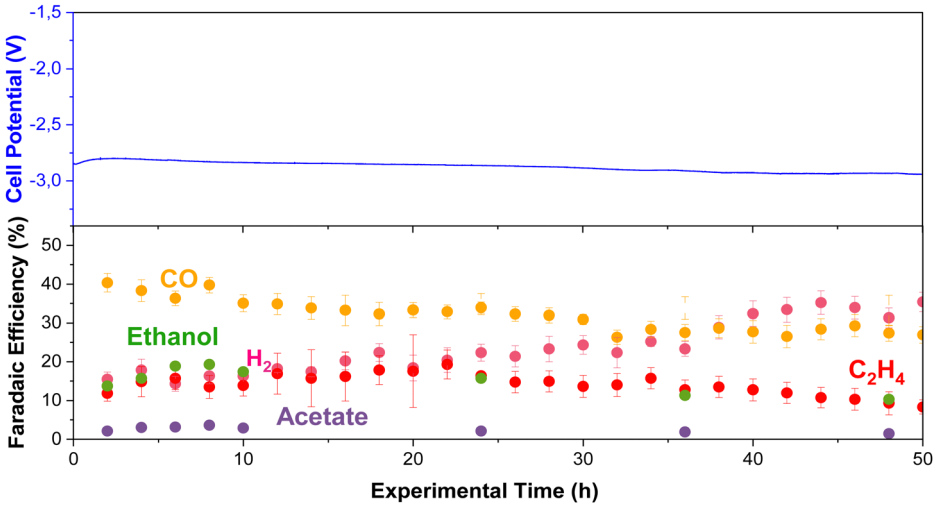

Fig. 9 Long-term stability test at 60 °C and 150 mA cm^{-2} using Cu-electrocatalyst. Experimental conditions: Cu-GDE (Cathode), MPIP-AEM, IrO_2 (Anode), CO_2 feeding rate 20 sccm, 0.1 M $KHCO_3$ as anolyte. Error bars represent the standard error of different injections and measured samples.

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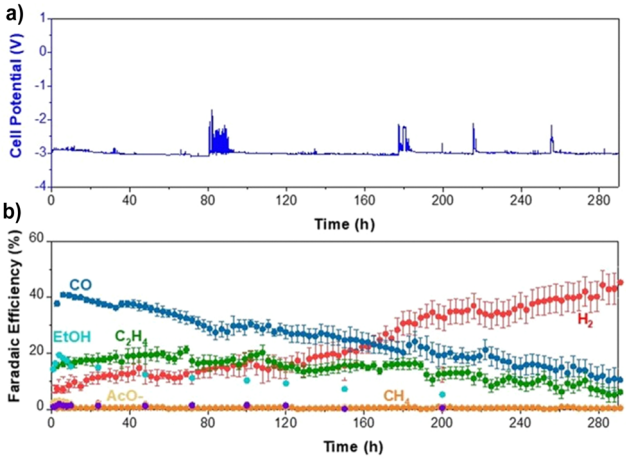

Fig. 10 290 hour CO_2 electrolysis stability test at 60 °C and 200 mA cm^{-2}. (a) Potential profile. (b) Faradaic efficiency. Operating parameters were sputtered Cu cathode with 10% be weight MPIP ionomer, MPIP membrane, IrO_2 on carbon anode and 0.1 M $CsHCO_3$ as an electrolyte. 100 mbar back pressure on the cathode was used as well. Error bars represent the standard error of different injections and measured samples.

$CsHCO_3$, as it is three times more soluble than $KHCO_3$.[42] Thus, by using a more soluble salt in Cs, a slight backpressure of 100 mbar on the cathode, and a 10% by-weight MPIP ionomer, we were able to slow the H_2 selectivity increase over time such that we were able to operate at 200 mA cm^{-2} for 290 hours at which point the H_2 reached 45% as shown in Fig. 10. This shows the importance of small details in optimizing performance.

Conclusions

We conducted a systematic study to investigate the effects of temperature on zero-gap CO_2 electrolysis to understand how this parameter influences selectivity, activity, and overall performance. We found that raising the cell temperature both increased the selectivity towards CO_2R over HER and concomitantly decreased the cell potentials due to enhanced kinetics. Additionally, we observed a significant shift of selectivity toward CO at temperatures above 50 °C (FE > 40%), which we associated with weakening the binding strength of intermediate CO to the Cu surface, favoring its desorption over C–C coupling reactions. Importantly, operation at elevated temperatures suppressed concomitant H_2 evolution (only 15% at 300 mA cm^{-2} at 80 °C) by mitigating against GDE flooding at high current densities to facilitate increased CO_2R durability. As a final point, we emphasize the importance of proper heating control and measurement when conducting such experiments, as differences in the heating methods might mislead the evaluation of the temperature effects during CO_2E cell testing. The insights obtained in this work highlight the benefits of conducting CO_2R at elevated temperatures under industrially relevant conditions. Strategies to optimize selective electrocatalysts, effective GDEs, and durable AEMs are needed to ensure that future enhancements can assist in overcoming the current limitations in the field. Such tests can only be conducted when an AEM is available that has both facile water dynamics and alkali stability at temperatures above 60 °C.

Author contributions

Carlos A. Giron Rodriguez carried out most of the electrochemical measurements, conducted data analysis, and was the lead writer of the manuscript with the input of all the other co-authors. Nishithan C. Kani did long-term stability and concomitant analysis. Asger Moss built the reaction set-up (heating oven) and contributed to conceiving and designing the research. Bjørt Oladottir Joensen did measurements to determine cathode potential and wrote the corresponding part. Sahil Garg helped with data analysis and performed experiments on CO reduction at elevated temperatures. Wanyu Deng conducted the ATR-SEIRAS experiments. From the University of Surrey Terry Wilson synthesized the AEMS that John Varcoe conceived and John contributed during the revision of the manuscript. Finally, Ib Chorkendorff and Brian Seger conceived the project, oversaw its development, and helped edit the overall manuscript.

Conflicts of interest

There are no conflicts to declare.

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