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Unraveling the single-atom electrocatalytic activity of transition metal-doped phosphorene†

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The development of single-atom catalysts (SACs) for chemical reactions of vital importance in the renewable energy sector has emerged as an urgent priority. In this perspective, transition metal-based SACs with monolayer phosphorous (phosphorene) as the supporting material are scrutinized for their electrocatalytic activity towards the oxygen reduction reaction (ORR), oxygen evolution reaction (OER), and hydrogen evolution reaction (HER) from first-principle calculations. The detailed screening study has confirmed a breaking of the scaling relationship between the ORR/OER intermediates, resulting in various activity trends across the transition metal series. Groups 9 and 10 transition metal-based SACs are identified as potential catalyst candidates with the platinum single atom offering bifunctional activity for OER and HER with diminished overpotentials. Ambient condition stability analysis of SACs confirmed a different extent of interaction towards oxygen and water compared to pristine phosphorene, suggesting room for improving the stability of phosphorene *via* chemical functionalization.

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1. Introduction

Endeavors to curb catalyst loading without dwindling the activity has been a far-reaching goal in the area of heterogeneous catalysis.^{1–3} Attempts towards the realization of this goal, along with the development of nanotechnology, have explored catalysts of different dimensions such as nanoclusters, nanowires, and 2D sheets. The most recent decades have witnessed the emergence of single-atom catalysts (SACs), whereby the atomically dispersed metals catalyzed the reactions and became quintessential examples for the green chemistry concept of “atom economy”.^{4,5} A vast number of categories of SACs have been explored so far by varying the supports, such as metal oxides, two-dimensional nanosheets, clusters, and metal organic frameworks.^{6–9} Apart from the stabilization effects of single atoms anchored on the supports, SACs offer extraordinary catalytic properties owing to their compelling electronic,

structural and chemical properties that arise from the synergistic interaction between the support and metal atom. For example, Zhang and coworkers demonstrated that Pt atoms uniformly dispersed over an iron oxide (Pt₁/FeO_x) support exhibited CO oxidation activity higher than their nanoparticle counterpart.¹⁰ This revolutionary study was followed by the discovery of a number of M/NO_x (M = Ir, Pt, Au; N = Ti, Fe, Ce) SACs identified as potential catalysts for the water-gas shift reaction.^{11–13} In order to address the expeditious energy crisis confronted by the modern world, the development of efficient electrocatalysts for the reactions of energy storage (such as the oxygen reduction reaction (ORR), CO₂ reduction reaction (COR), oxygen evolution reaction (OER) and hydrogen evolution reaction (HER)) is an area of utmost research interest. Since the primary catalysts used at an industrial level for these reactions are noble expensive metals, such as Pt, Pd, Ir and Rh, the evolution of SACs has set off enormous attempts to reduce catalyst loading by retaining/improving the activity of the one-atom based catalysts. So far, the combined experimental and theoretical studies have unraveled a number of transition metal-based SACs with pristine and vacant graphene-type materials and nanoparticles.^{14–17} Continuous research is being carried out to disclose novel SACs with optimal metal-support combinations for excellent activity.^{18–20}

Few-layer bulk phosphorus, particularly monolayer phosphorene has been proposed as a material of immense interest owing to the exquisite properties, such as its unique anisotropic electronic and transport properties, high mobility, turn off ratio and tunable direct band gap.^{21–23} These properties have enabled phosphorene to be employed for a diverse set of applications, such as gas sensing, transport devices and optoelectronics.^{24–26}

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The major obstacle in the practical utilization of phosphorene is its instability at ambient conditions, the origin of which has been attempted to be understood at an atomic scale.^{27,28} However, recent studies have been able to synthesize air-stable few-layer black phosphorous (BP) by layer-by-layer thinning of the bulk black phosphorous and by pulsed laser exfoliation.^{29,30} Such air-stable phosphorene nanosheets have also been identified with a preferred catalytic activity towards electrochemical reactions. In a pioneering study by Zhang and co-workers, few-layer BP nanosheets were identified as potential catalysts for OER, while retaining structural stability over 1000 potential cycles.³¹ The linear relation between the reduction in the thickness of the BP nanosheets and the electrocatalytic activity suggested by the authors recommended the plausibility of monolayer phosphorene as an efficient electrocatalyst. However, Xue *et al.* showed through a theoretical study that the enhanced catalytic activity for ORR or OER could be derived from phosphorene only by inducing modifications, such as vacancies and hetero-atom doping.³² Therefore, the selective functionalization of phosphorene draws considerable attention as it can deliver significant electrocatalytic activity by tuning the interaction with reactive intermediates, as well as enhancing the stability by modifying the electronic structure.

Motivated by the aforementioned ideas, we attempted to investigate the activity of transition metal-doped phosphorene SACs (TM-Ph) towards major electrocatalytic reactions, such as ORR, OER and HER based on first-principles calculations. The final transition metals from the 3d (Fe, Co, Ni, Cu), 4d (Ru, Rh, Pd, Ag) and 5d (Os, Ir, Pt, Au) series were considered owing to their catalytic applications for various processes at the bulk level. The structural, electronic and thermodynamic aspects of the catalytic activity were systematically studied and an activity screening among the considered catalyst species was carried out. The origin behind the activity trends observed was scrutinized to derive suitable activity descriptors and related

parameters. The active candidates were further analyzed for their stability under ambient conditions by an atomistic approach. The results are expected to provide useful thumbnails for future efforts in enabling phosphorene as an efficient support for SACs.

2. Computational model and details

The density functional theory (DFT) calculations were carried out using VASP (Vienna *Ab Initio* Software Package).³³ Gradient corrected approximation using Perdew–Burke–Ernzerhof (GGA–PBE) exchange–correlation functional was employed with the projected augmented wave method (PAW) to describe the periodic boundary conditions.^{34,35} The study by Hashmi and Chan reported on the exactness between the magnetic ground state prediction between DFT–GGA and DFT–U for the 3d TM–Ph systems.³⁶ Hence, we have avoided considering the Hubbard calculation in our study. A kinetic-energy cutoff of 470 eV with a *K*-point grid of $5 \times 5 \times 1$ for the Brillouin zone sampling was used for the energy calculations, and $15 \times 15 \times 1$ was used for the electronic structure calculations. The dispersion interactions were accounted for by using the DFT–D3 method.³⁷ All calculations were spin polarized and without any symmetry constraints. An implicit solvation model implemented in VaspSol was used to simulate the electrolyte conditions.³⁸ Crystal orbital Hamilton population (COHP) analysis was carried out using the Lobster program.³⁹

A $4 \times 4 \times 1$ supercell with 48 atoms was used to model phosphorene with the transition metals substitutionally doped by replacing one phosphorous atom at the bridge site, constituting an atomic doping percentage of 2.708% (Fig. 1). The doped TM atoms were observed to possess variable geometric features across the series, which are given in Table S1.† While the bond lengths of TMs with P atoms (TM–P1/P4) along the zigzag direction remained close to the pristine values (2.17–2.47

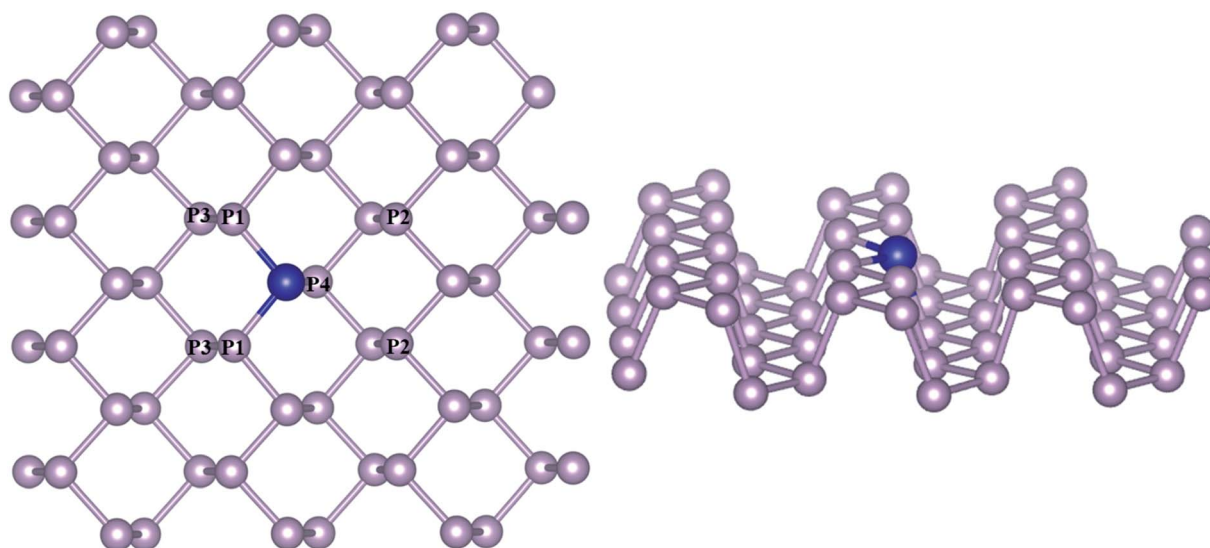


Fig. 1 Transition metal-doped phosphorene model (a) top view and (b) side view. P atoms differing in coordination with respect to the TM atom are labeled. Blue and violet colors represent the TM and P atoms, respectively.



Å), the average TM–P bond length along the armchair directions (TM–P2) differed considerably across the TM series (2.5–3.6 Å), with a closeness observed for the same group metals. The incorporation of TMs was found to be associated with a diminished buckling as compared to pristine phosphorene, which was determined from the interplanar distance (P1–P3) between the outer and basal planes. Of all the TM–Phs considered, only the Fe and Ni-doped systems were magnetic with a total magnetic moment of 1 μ_B .

3. Results and discussion

The energetics of the phosphorene monovacancy formation, the prerequisite factor for TM doping was analyzed by calculating the formation energy as follows:

$$F_E = E_{\text{Ph-Vac}} - E_{\text{Ph}} - E_P \quad (1)$$

where $E_{\text{Ph-Vac}}$ is the energy of the vacancy-induced phosphorene, E_{Ph} is the energy of the pristine phosphorene monolayer and E_P is the chemical potential of phosphorous calculated from bulk phosphorous. The calculated formation energy of the monovacancy in phosphorene was 2.094 eV, which is less than the

single vacancy formation energy reported for graphene.⁴⁰ Therefore, the induced vacancy can be expected to be experimentally achieved *via* advanced methods like the chemical exfoliation technique. This is followed by an analysis of the TM binding energy, which is calculated as follows:

$$E_{\text{BE}} = E_{\text{TM-Ph-Vac}} - E_{\text{Ph-Vac}} - E_{\text{TM}} \quad (2)$$

where the terms from the left to right indicates the energy of the TM-doped vacancy-created phosphorene, vacant phosphorene and the chemical potential of the TM atom determined from its bulk counterpart, respectively (Fig. 2a).

All TM–Ph based systems except Ag–Ph have been found to bind strongly, and hence possess the required thermodynamic stability to be useful as catalyst materials. The structural integrity and tolerance towards electrochemical conditions are crucial parameters in determining the long-term activity of electrocatalysts, and hence to be addressed in detail. Within this consideration, the relative binding energy of TM-dimer on phosphorene was determined to analyze the plausibility of aggregation of the doped TMs under a TM-rich condition (Fig. 2a: red panel), and compared with a single-atom binding scenario. All TM-dimers were found to be weakly binding

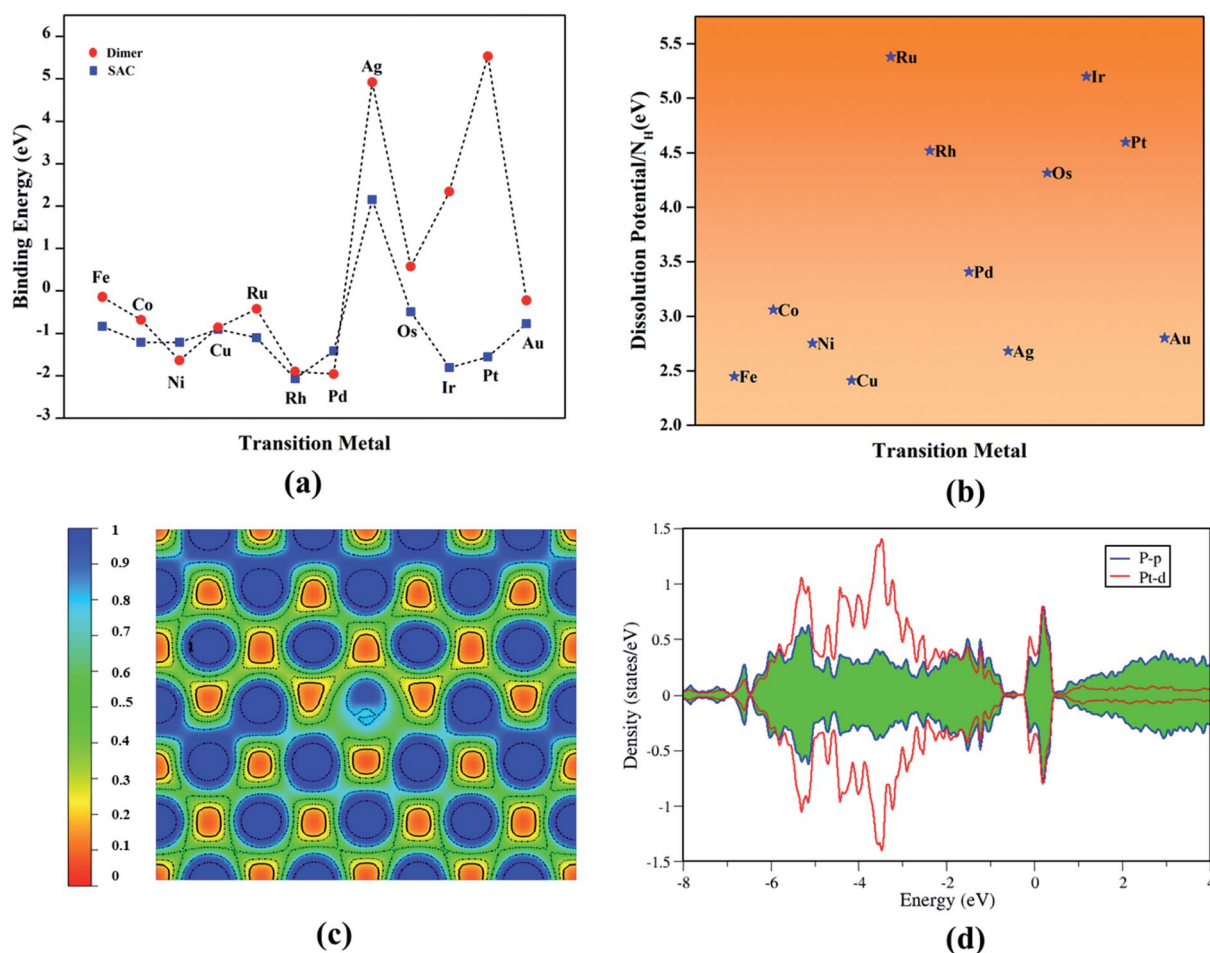
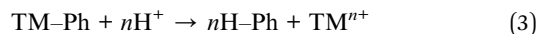


Fig. 2 (a) Binding energy trends of transition metal single atoms and dimers. (b) Dissolution potential of SACs per number of protons (N_H). (c) Electron localization function and (d) Projected density of states of prototype Pt–Ph system.



compared to the single TM atom, except for Ni, Pd undergoing dissociation, suggesting a reduced tendency towards aggregation for the SACs. Furthermore, the sustainability of the TM–Ph SACs under electrochemical conditions was scrutinized by the dissolution potential of TMs calculated from replacing the TM atoms with protons, followed by the work of Singh *et al.*⁴⁴ as the free energy of the reaction:



with $n\text{H-Ph}$ indicating the TM vacancy occupied by n hydrogen atoms according to the standard oxidation state of the metal. The observed positive dissolution potentials (Fig. 2b) per number of protons subjected to the oxidation state of the metals for all TM–Ph systems conveyed that they were not prone to dissolution during subsequent protonations. Moreover, the minimum pH value of less than 1 required for the TM–Ph SACs to resist the dissolution (Table S2[†]) also confirmed the stability under a higher acidic and alkaline pH range. A topological analysis was carried out based on the electron localization function (ELF) (Fig. 2c) to understand the nature of bonding between the TM and surrounding P atoms. The localization of the electron pairs at the atomic centers was revealed by the

maximum ELF value of close to 0.40, which infers a dominance of ionic bonding character. The projected electronic density of states for all TM–Ph systems was inspected and that of the Pt–Ph candidate, for which a high binding energy and dissolution potential were observed, is shown in Fig. 2d as a prototype. The strong hybridization observed between the d-orbitals of TMs and the p-orbitals of the coordinated P atoms near to the Fermi level explains the origin of the electronic stability of the TM–Ph systems. A similar interaction pattern was also observed for the remaining TM–Ph SACs, and is represented in Fig. S1.[†] Hence, the stability of the TM–Ph based SACs was confirmed, and the TM–Ph based SACs were further screened for their catalytic activity.

3.1 Oxygen reduction and evolution reaction

Owing to the similarity between the reaction intermediates involved, we attempted to first study the ORR characteristics of the TM–Ph SACs and utilize the relevant aspects to describe the OER activity trends. The adsorption configuration of the ORR intermediates is given in Fig. 3a–e. ORR was initiated by O_2 adsorption, the details of which are given in Table S3.[†] The absence of local minima with O_2 physisorbed on the TM atoms ensured favorable binding to the single atom center, rather than

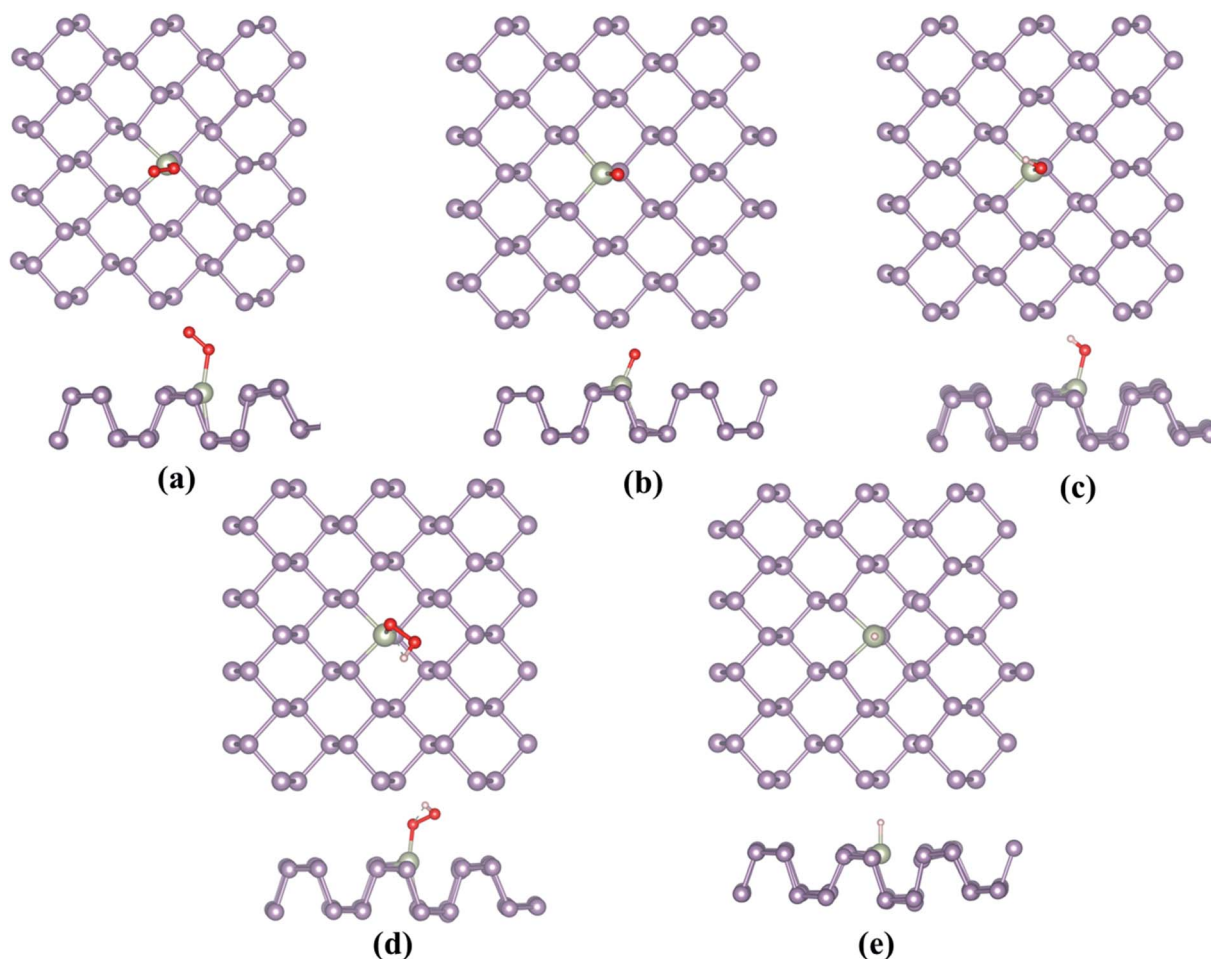


Fig. 3 Most stable adsorption configurations of the ORR/OER/HER intermediates, (a) O_2 , (b) O^* , (c) OH^* , (d) OOH^* and (e) H^* .



the P atoms where the physisorbed O₂ configuration was observed, which is discussed in a later section. A characteristic linear adsorption mode was observed for all TM-Ph SACs with an average vibrational frequency of 1100–1300 cm⁻¹, except for Pd-Ph adsorbing in a di-sigma mode. The O₂ binding was highly localized at the metal centers in TM-Phs with a lesser O₂ binding strength compared to the P atoms of the pristine phosphorene, among which the highest interaction was evidenced for Fe-Ph with a binding energy of -1.77 eV. The corresponding projected density of states (PDOS) analysis (Fig. S2†) of Fe-Ph showed a strong hybridization between the p states of O₂ and the d-states of iron, retaining a magnetic moment of 3 μ_B. This indicated an additive coupling between Fe and the triplet O₂ magnetic moments. The localization of O₂ binding on the transition metal atoms was further confirmed by the relatively higher P-O* bond lengths (>3 Å) compared to the TM-O* bond length (<2.5 Å), as shown in Fig. S3.† Top site O* adsorption occurs on the TM centers with an exception for the end series metals, such as Ni, Cu, Pd, Ag and Au, where a bridge type adsorption with P2 atoms occurs. OH*, OOH* and H* adsorptions also occurred in the top mode across all the SACs, which follows their typical adsorption configurations over the (111) facets of the transition metal surfaces. The local adsorption-induced structural changes quantified in terms of the deviation of the TM-P bond lengths in Fig. S4† confirmed the structural deviations within a 0.1 Å change, except for Cu- and Au-based SACs, for which relatively large bond length variations were present. For Ni and Pd, a strong H* binding caused the TM to translate towards the lower layer of TM-Ph, and hence led to a distortion of the system. Therefore, these (Ni and Pd) SACs were not considered for studying the HER activity.

Since ORR/OER is a multistep reaction with intermediates O*, OH* and OOH* involved, it is preferable to search for an activity descriptor especially based on one intermediate so that a parametric space for the catalytic activity can be simplified in the number of degrees of freedom and a general trend can be expected. Having this as the objective, we investigated the scaling relationship between the adsorption free energies of the intermediates, which were attributed to the fundamental origin for the diminished experimental onset potential for ORR (Fig. 4). The adsorption free energies were calculated with reference to the water from the following equations:

$$\Delta G_{O^*} = G_{O^*} - G^* - [G_{H_2O} - G_{H_2}] \quad (4)$$

$$\Delta G_{OH^*} = G_{OH^*} - G^* - [G_{H_2O} - 1/2G_{H_2}] \quad (5)$$

$$\Delta G_{OOH^*} = G_{OOH^*} - G^* - [2G_{H_2O} - 3/2G_{H_2}] \quad (6)$$

with the zero-point energies and entropies calculated from the harmonic oscillator approximation, along with the DFT calculated binding energy. The interdependency between the intermediate adsorption energies was analyzed *via* the scaling relationships (shown in Fig. 4), which were identified to be a pivotal factor in determining the electrocatalytic activity.^{14–17}

A surprising observation was the consistency in the linear scaling of $\Delta G_{OOH^*} = 0.96\Delta G_{OH^*} + 3.13$ with a large determination coefficient between OH* and OOH*, as seen for the transition metal surfaces. Conversely, there was almost negligible scaling found for OH* *vs.* O* and OOH* *vs.* O*. This suggested a breaking of the scaling relationships across the TM-Ph series, which was a sufficient reason for the catalytic trends. The involvement of nearby phosphorous atoms in bonding with the O atoms could be expected to provide a difference in the binding of ORR intermediates by the SACs and hence gives rise to the nonlinear scaling relationships, while the similar mode of adsorption (top site) sustained the relationship between OH* and OOH*. This observation immediately denied the possibility of considering the O* binding energy as an activity descriptor, which had been considered in many studies. To identify the chemical origin of breaking the scaling relationships, we carried out a projected crystal Hamilton population analysis (pCOHP) for all intermediates, which was recently verified to be an important activity descriptor for computational screening studies (Fig. 5).⁴² The bonding and antibonding populations were represented on the right and left sides of the plot, respectively. The depicted O* pCOHP involved an average of the TM-O and P-O bonding for the Ni, Cu, Pd, Ag, and Au based SACs as the O* occupied a bridging position in these systems. The observed discrepancies in the adsorption free energy, and hence the scaling relationship, could be ascribed to the relatively higher variation in the spread of the bonding and antibonding populations for the TM-O* binding as compared to the TM-OH* binding. Another noteworthy aspect arising from the pCOHP analysis was that all TM-Ph catalysts possessed

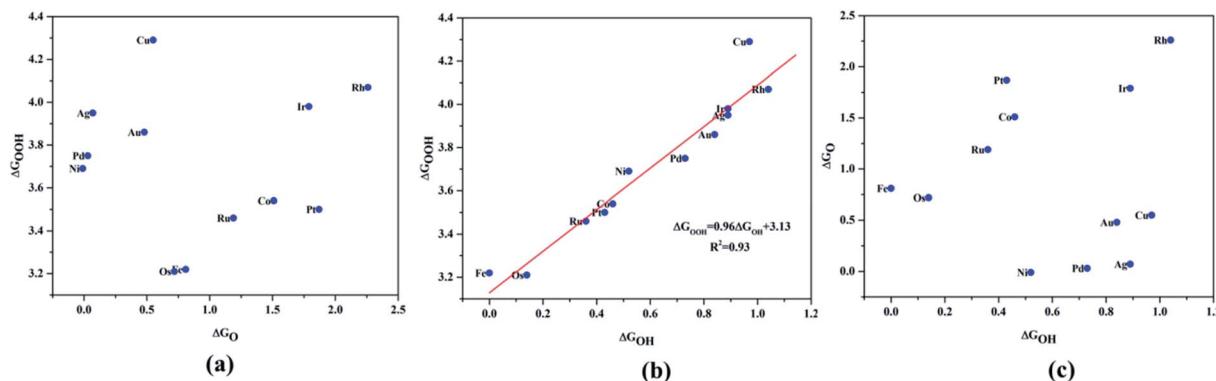


Fig. 4 Scaling relationship of the ORR/OER intermediates, (a) ΔG_{O^*} vs. ΔG_{OH^*} , (b) ΔG_{OOH^*} vs. ΔG_{OH^*} , (c) ΔG_{O^*} vs. ΔG_{OOH^*} .



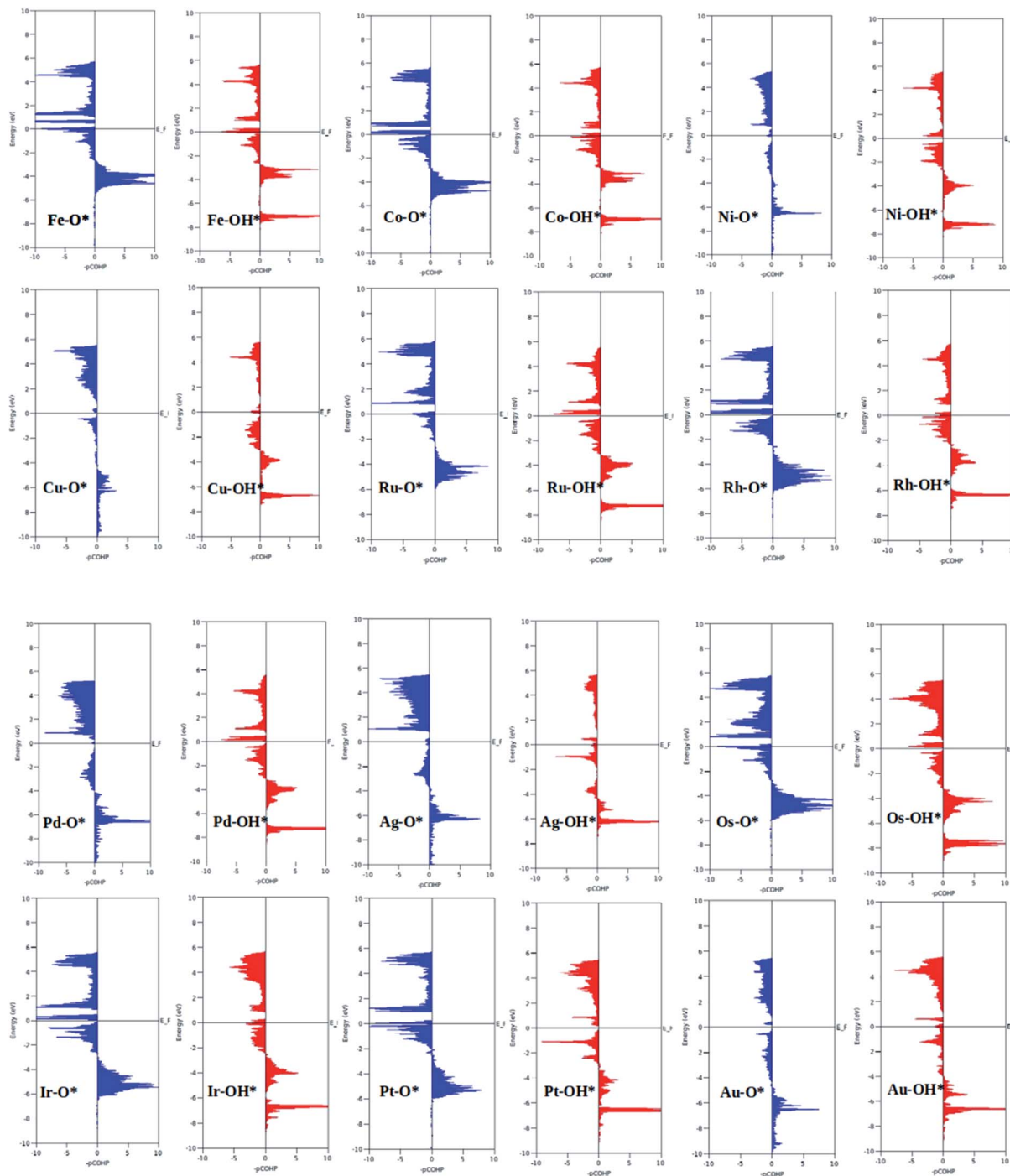


Fig. 5 Projected crystal Hamiltonian population analysis of the O* and OH* intermediates adsorbed on TM-Ph SACS.

a slightly higher population for the TM-O* bonding energy levels than the antibonding energy levels at the valence band. This indicated an optimum bonding scenario that reduced the possibility of strong O* binding-caused catalytic site poisoning, a major drawback of the conventional transition metal-based catalysts. Furthermore, a plot of the integrated COHP (ICOHP) up to the Fermi level against the Mulliken's charge transfer⁴³ revealed a higher order linearity between the charge transferred to OH* compared to O* (Fig. S5†). O* binding is associated with

a larger charge span of 0.4–1.2|e|, which is still far from the valence satisfying criteria of 2e⁻. Conversely, OH* binding features a lower charge fluctuation (0.85–1.06|e|), which is consistent with the valency requirement of 1e⁻. This difference in the strengths of the covalent character has been previously observed for OH* and OOH* in MN₄ complexes by Calle-Vallejo *et al.*,⁴⁴ causing a breaking of the scaling relationships in vacuum and is retained under implicit solvent consideration. Since our study involved an implicit solvation consideration,





Fig. 6 Free energy diagrams of ORR and OER along the TM–Ph SAC series constructed at 1.23 V.

the differences in the covalence of the ORR intermediates emerged as a parameter of interest for the activity enhancement by tuning the metal center. The free energy diagrams for ORR (Fig. 6) constructed at the equilibrium potential of 1.23 V indicated that there was a non-uniform trend in the energetics of ORR over different TM–Ph SACs.

The TM–Phs with 3d series elements Fe, Ni and Cu, along with the end-lying Pd, Ag and Au of the 4d and 5d series featured OH* formation as the rate-determining step. On the contrary, Co, Ru, Os, Ir and Pt-based TM–Phs possessed H₂O formation as the rate-limiting step. Rh–Ph emerged as the only system with OOH* formation as the rate-limiting step. Therefore, it was found that the stability of OH* on the catalyst surface can be considered a crucial parameter in determining the catalytic trends among the SACs. From the free energy screening, Ir–Ph was the best ORR catalyst with an overpotential of 0.34 V calculated using the computational hydrogen electrode model (CHE) proposed by Nørskov *et al.*,⁴⁵ which was followed by Rh and Co with overpotentials of 0.39 and 0.77 V, respectively. It is interesting that Ir–Ph manifested an almost

constant (although non-zero) ΔG (0.30, 0.37, 0.33 and 0.34 eV) for all four protonation steps of ORR, which is an adequate feature of an ideal catalyst. The lowest overpotential value of 0.34 eV reinforced the scaling relationship–driven activity trend by its excellent resemblance with the minimum thermodynamic overpotential of 0.33 V derived from the linear scaling between OH and OOH [(1.23 – (–4.92 + 3.12)/2) V = 0.33 V]. The obtained overpotentials for the active TM–Ph catalysts (Ir–Ph: 0.34 V and Rh–Ph: 0.39 V) were less than the well-known ORR catalyst periodic Pt(111) surface with a value of 0.45 V (ref. 45) (given in Fig. 7a), confirming that these TM–Ph catalysts greatly improved the ORR activity along with a considerable minimization of metal loading. Compared to ORR, the OER activity followed a different trend across the SACs with a uniform rate-determining step of OOH* formation. The highest activity was observed for Pt–Ph for which a free energy change of 0.40 eV was required for the OOH* formation from O*, which was followed by Rh–Ph with a corresponding value of 0.58 eV. The highest activity of Pt–Ph can be attributed to the ΔG_{O^*} value of 1.87 eV lying closest to the mean value of ΔG_{OH^*} and ΔG_{OOH^*} . This is an



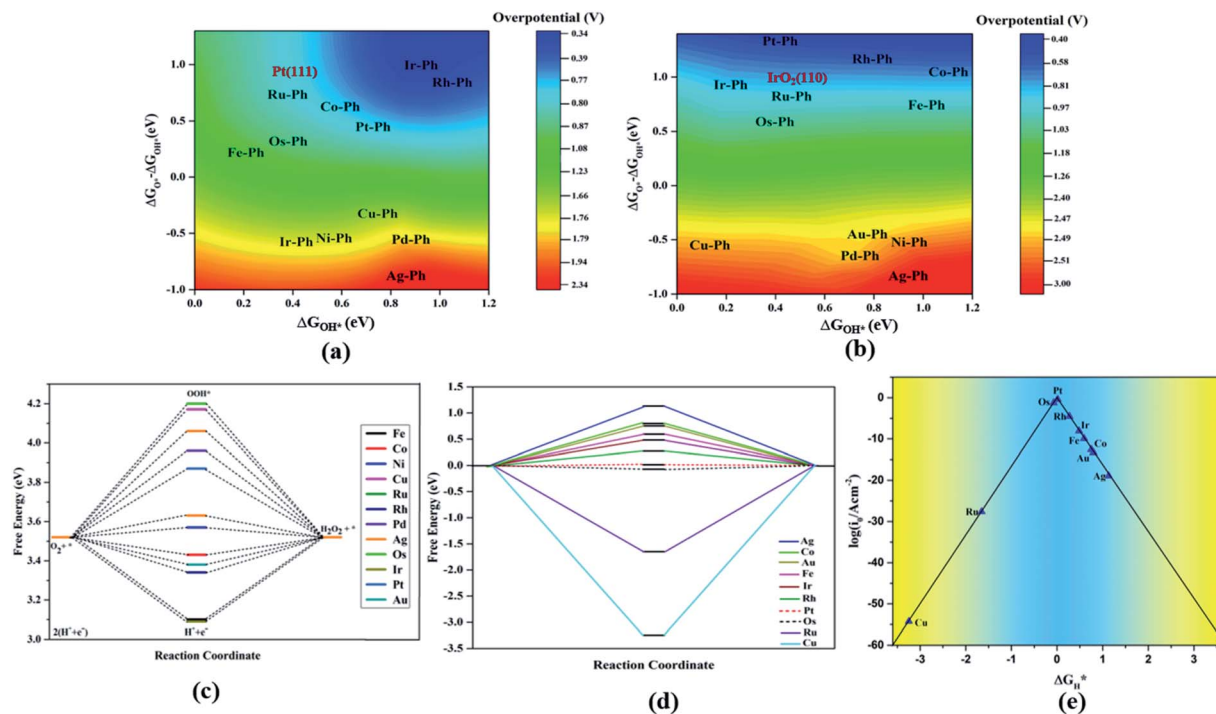


Fig. 7 (a and b) Two-dimensional ORR (a) and OER (b) activity volcano plots. The activities of the Pt(111) and $\text{IrO}_2(110)$ reference systems calculated at the GGA–RPBE level are shown in brown. (c) Free energy analysis of $2e^-$ reduction by TM–Ph SACs at 0.7 V. (d) HER free energy diagram of TM–Ph SACs. (e) Current density volcano plot of HER vs. ΔG_{H^*} .

essential criterion for an efficient catalyst for which either the conversion from O^* to OOH^* or OH^* to O^* can be the rate-determining step.⁴⁶ The dominance of the single Pt atom over Ir or Ru, for which the oxide counterparts are well-known OER catalysts for their high activity, is noteworthy. For example, the well-known OER catalyst periodic $\text{IrO}_2(110)$ surface featured an overpotential of 0.51 V (ref. 46) (given in Fig. 7b), which could be outperformed by Pt–Ph SAC with a diminished overpotential by 0.11 V.

For both ORR and OER, high fluxionality was observed between the free energy of the elementary reaction steps across the SACs with a notable uniformity in the end-lying metal center in the TM series (Cu, Ag, Au) exhibiting the lowest activity. This can be ascribed to their high d electrons causing a strong binding towards the intermediates, introducing an energetically uphill trend in the corresponding reactions. This can be also understood from the alignment of these TM–Ph SACs in the strong charge-transferring category during the TM–O* interaction from the Mulliken's charge analysis in Fig. S5(a).[†] Activity contour plots of the TM–Ph SACs for the ORR and OER are represented in Fig. 7a and b with ΔG_{OH^*} and $\Delta G_{\text{O}^*} - \Delta G_{\text{OH}^*}$ as the reference parameters, as the identified scaling between OOH^* and OH^* reduces the dependency of the ORR/OER activity from four ΔG values into two. From the volcano plots, it is evident that the Ir–Ph, Rh–Ph, Pt–Ph and Os–Ph SACs occupied the highest activity regions and could be considered as potential catalysts. It is also noteworthy that the less stable Ag–Ph and Pd–Ph SACs featured less electrocatalytic activity, as identified from the activity volcano plot.

To elucidate the thickness dependence on the electrocatalytic activity of TM–Ph-based SACs, we calculated the activity of Ir–Ph and Pt–Ph for ORR and HER, respectively, up to a thickness (Fig. S6 and Table S4[†]) of 3 monolayers (3 MLs). Both systems showed a decrease in the activity with higher overpotentials for 2 ML and 3 ML thicknesses than the 1 ML counterparts, suggesting an inverse relation between the ORR activity vs. thickness. However, the HER activity underwent a comparatively lower (by 1 V) and irregular variation with thickness.

A competitive pathway for the $4e^-$ ORR with H_2O formation is the $2e^-$ reduction resulting in H_2O_2 formation. Simultaneously, the electrochemical synthesis of H_2O_2 is an important area of industrial importance.^{47–49} Owing to the similarity in intermediates, we investigated the $2e^-$ oxygen reduction activity of TM–Ph SACs by a free energy analysis with the OOH^* formation and H_2O_2 formation as the two elementary steps. Fig. 7c represents the free energy diagram at the equilibrium potential of 0.7 V of the $2e^-$ reduction. The ORR/OER-active TM–Ph SACs did not show prominent activity towards the $2e^-$ reduction, which diminished the plausibility for the competitive $2e^-$ vs. $4e^-$ reduction pathways. Instead, the 3d series-based TM–Ph SACs (such as Ni–Ph and Co–Ph) emerged as active catalysts with a curtailed energy requisite of 0.05 and 0.09 eV that led to limiting potentials of 0.65 and 0.61 V, respectively. This uncovered a catalytic activity that is equivalent to the performance of the state-of-the-art $2e^-$ reduction catalyst PtHg.⁵⁰ There is variation in the activity-determining reaction from the OOH^* formation to OOH^* protonation along the TM–



Ph series that occurred due to the weak and strong binding of OOH^* to the metal center, respectively.

3.2 Hydrogen evolution reaction

Apart from the oxygen reduction/evolution reactions, we further investigated the hydrogen evolution activity of the TM-Ph SACs. Many of the recently developed SACs have offered prevalent catalytic activity towards HER. Since Ni-Ph and Pd-Ph have been identified to be structurally unstable during H^* adsorption, we avoided them for the HER activity study. Since ΔG_{H^*} is the only activity descriptor involved, we considered a Heyrovsky mechanism for our study subjected to the fact that the protons were adsorbed into the catalytic center from the solution and later desorbed as H_2 .⁵¹ From the HER free energy diagram given in Fig. 7d, it was obvious that H^* formation was the rate-determining step for the majority of the SACs, whereas the RDS for Cu and Ru was the combination of the adsorbed H^* with protons to form H_2 . Here, Pt-Ph was the best SAC because it almost achieved the ideal efficiency with $\Delta G_{\text{H}^*} = 0.01$ eV. This was followed by Os-Ph with an overpotential of 0.07 V. A volcano plot of the exchange current density (a parameter which can be measured experimentally) versus the Gibbs free energy of adsorption is represented in Fig. 7e, where the current density was calculated by the following method proposed by Norskov *et al.*:⁵²

$$i_0 = \begin{cases} \frac{-ek_0}{1 + \exp\left[\frac{-G(\text{H}^*)}{k_{\text{B}}T}\right]} & \text{for } G(\text{H}^*) \leq 0 \text{ and } i_0 \\ \frac{-ek_0}{1 + \exp\left[\frac{G(\text{H}^*)}{k_{\text{B}}T}\right]} & \text{for } G(\text{H}^*) \geq 0 \end{cases} \quad (7)$$

where $G\text{H}^*$ is the free energy of the H^* adsorption, k_{B} is the Boltzmann constant, T is the temperature and k_0 is the rate constant, which has been set to be 1 because of the unavailability of experimental results for the TM-Ph SACs HER activity. Pt-Ph occupied the topmost position of the activity volcano, suggesting that it showed the highest HER activity. Hence, the activity screening showed that phosphorene-based SACs with transition metals belonging to groups 9 and 10 emerge as the most active catalyst candidates (except Pd) in consideration of all electrocatalytic reactions. From our study, Pt-Ph also emerged as a bifunctional catalyst that simultaneously offered the lowest Pt loading, together with excellent activity for OER and HER, and can be expected to be a pivotal material of catalytic interest.

3.3 Ambient condition stability of SACs

Phosphorene is known to have stability issues under ambient conditions that are caused primarily by its interaction with light, atmospheric oxygen and water/moisture. As many of the

TM-Ph SACs are characterized by their exceptional electrocatalytic activity, we attempted to investigate their ambient condition stability *via* studying the chemical degradation of pristine phosphorene from first-principle calculations. The ambient condition stability investigations were carried out for the most active catalysts (Ir-Ph, Rh-Ph, Pt-Ph, Os-Ph) for all electrocatalytic reactions, considering pristine phosphorene as the frame of reference. To begin with, we carried out *ab initio* molecular dynamics (AIMD) simulations of these SACs at 300 K with a time step of 2 fs for a time period of 10 ps. The fairly smooth free energy fluctuation during the trajectory evolution (Fig. S7†) confirmed the thermodynamic stability of these SACs. This was followed by a detailed investigation of the chemical interactions of SACs with its surroundings.

The mechanism of the O_2 -induced degradation has been studied in detail by Ziletti *et al.*, which is mediated by the breakage of the structural integrity of the phosphorene monolayer *via* the dangling oxygen atoms bound to P atoms.⁵³ We extended the three different reported O_2 interaction modes on phosphorene to the selected Ir-Ph, Pt-Ph, Rh-Ph and Os-Ph SACs, which are: (i) physisorbed O_2 ($\text{O}_2\text{-Phy}$), (ii) O_2 chemisorbed in a bridge fashion ($\text{O}_2\text{-Che}$), and (iii) dissociated O_2 resulting in the formation of two P-O dangling bonds ($\text{O}_2\text{-Dis}$) on the non-TM region of the TM-Ph systems. The structural details of the three different adsorption modes are given in Table S5.† Deviations in the magnetic moments from $2 \mu_{\text{B}}$, as observed for pristine phosphorene, in some of the TM-Ph SACs suggest different interactions in the physisorption mode. The average binding energies of the dangling O atoms for all TM-Ph SACs (except Pt-Ph) are lower than that of pristine phosphorene, implying two less stable (P-O*) configurations for the respective SACs and a diminished affinity towards oxidation. Furthermore, we investigated the free energy change associated with the transfer of the O_2 interaction *via* the discussed three modes (results are represented in Fig. 8). Although there is a thermodynamic plausibility for the movement of O_2 from the physisorbed to chemisorbed state, there is less exothermicity associated with the O_2 dissociation for the SACs in comparison to the scenario of pristine phosphorene, especially for the highest ORR-active candidate Ir-Ph with the least exothermicity observed. This observation is supported by the low O^* binding energy, which can be attributed to the relatively large occupation of the antibonding energy levels. These antibonding energy levels occur under the valence band region for the lowest binding Ir-Ph SAC, as determined from the pCOHP analysis of P-O interaction against pristine phosphorene (Fig. S8†). Transition metal states close to the Fermi level were previously revealed from the density of states analysis. These states can be expected to reduce the pure p orbital character for the P atoms belonging to the immediate ridges, resulting in less interaction with the dissociated O^* atoms. These results introduce a future platform where the ambient condition stability can be tuned *via* chemical modification strategies by varying the composition, doping percentage and synergistic methods with protective measures as well.

Another important degradation pathway for phosphorene is *via* the interaction with H_2O . Previous reports showed that



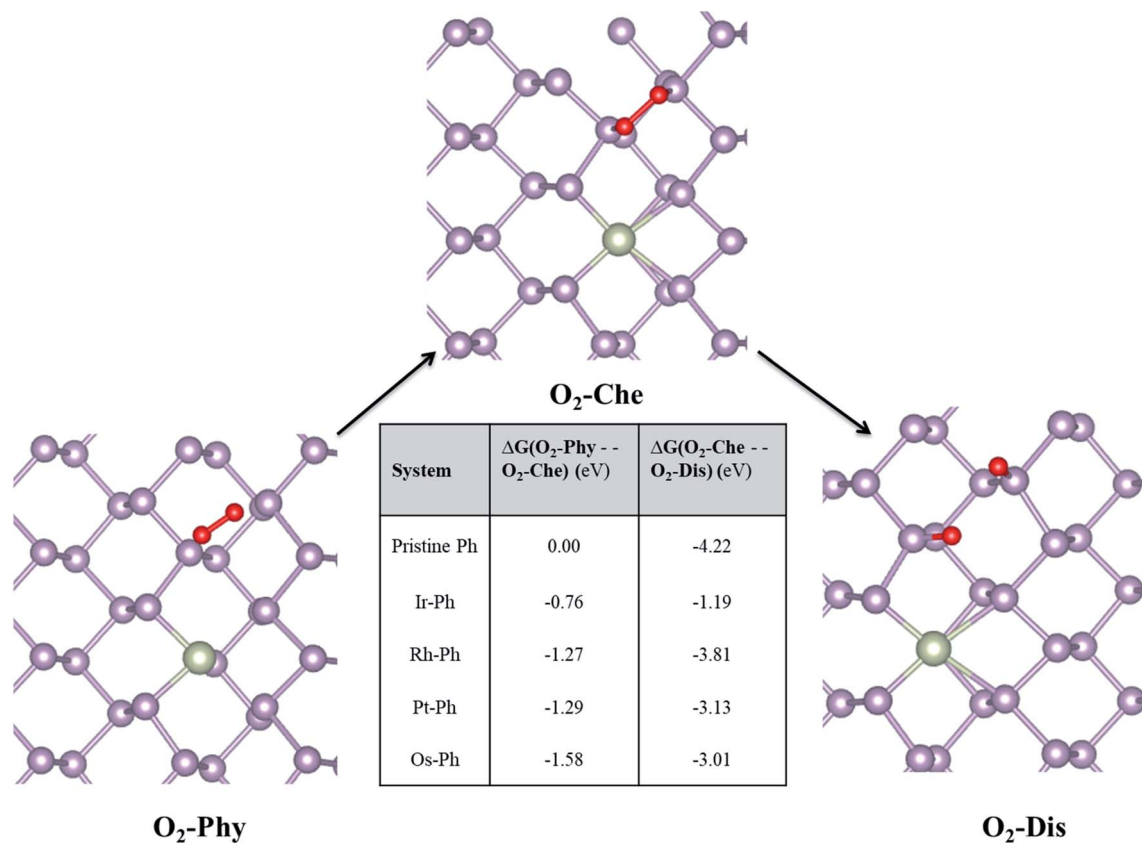


Fig. 8 Free energy analysis of O_2 interaction with TM-Ph SACs. Pristine phosphorene is the reference system. $\text{O}_2\text{-Phy}$, $\text{O}_2\text{-Che}$, and $\text{O}_2\text{-Dis}$ represent the physisorbed, chemisorbed and dissociated oxygen molecule, respectively.

although phosphorene displays fairly strong hydrophobicity towards H_2O , the oxygen-bonded phosphorene is vulnerable to water-catalyzed oxidation.^{54,55} Herein, we analyzed the

interaction of water with the O^* -bonded TM-Ph systems, using pristine phosphorene as the reference system. The charge density difference analysis of the H_2O interaction with O^*

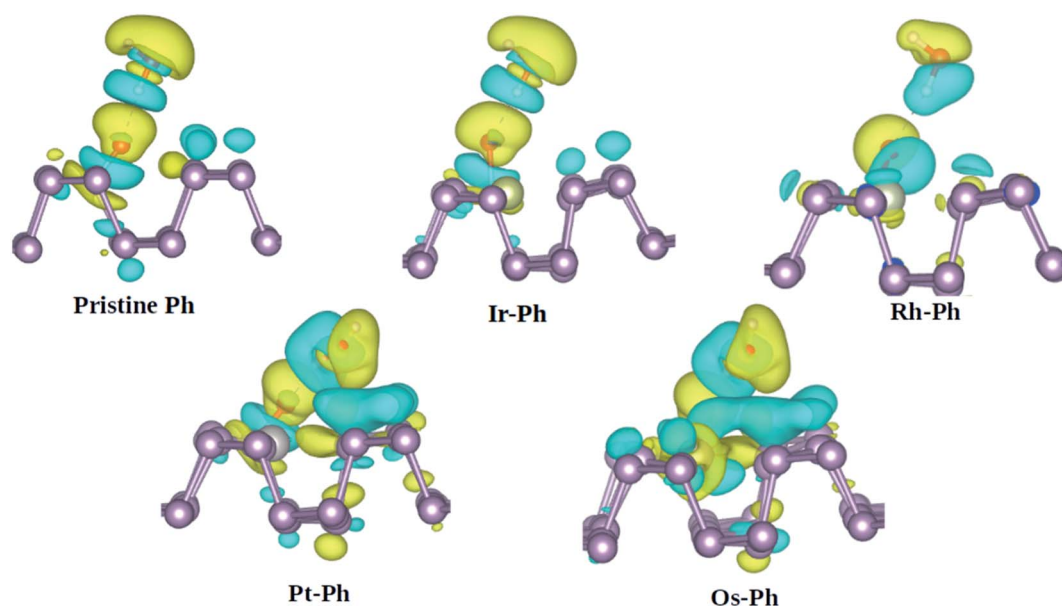


Fig. 9 Charge density difference analysis of H_2O with the O^* -bonded TM-Ph SACs and pristine phosphorene. Yellow and cyan colors represent the positive and negative isosurfaces, respectively, at 0.003 eV.



bonded to the P atoms of the TM–Ph systems is represented in Fig. 9.

From Fig. 9, it can be understood that although Ir–Ph showed a similar hydrogen bonding interaction to that of pristine phosphorene, Rh–Ph showed less interaction with the charges mainly located within water and the O* species only. The interactions of both Pt–Ph and Os–Ph with H₂O involved contributions from the TM atoms as well, and spread over a larger number of P atoms across the upper plane of phosphorene. This indicates that, except for Ir–Ph, all TM–Ph systems exhibited a different extent of interaction with H₂O. Rh–Ph exhibited a diminished charge transfer, which suggests the plausibility of controlling the H₂O affinity of oxidized phosphorene *via* chemical modification.

4. Conclusion

We systematically carried out a study of the catalytic activity of transition metal-doped single atom catalysts (TM–Ph SACs) towards important catalytic reactions, such as ORR, OER and HER. The stability analysis *via* binding energy, dissolution potential and electronic structure analysis confirmed that all TM–Ph SACs (except Ag–Ph) exhibited excellent stability in withstanding the electrochemical conditions. We also observed a breaking of the scaling relationship between the O* and OH* free energy of adsorption, which arises from the differences in the covalent characters attained by the respective species across the SAC series. The free energy analysis suggest that Ir–Ph:Rh–Ph, Pt–Ph:Os–Ph and Pt–Ph:Co–Ph are the most active catalyst systems towards ORR, OER and HER, respectively. The 3d series based SACs, Ni–Ph and Co–Ph, were also found to be active catalysts for 2e[−] ORR. The ambient condition stability analysis focused on the interactions of TM–Ph SACs with O₂ and H₂O, suggesting that these interactions can be tuned by incorporating TM atoms into pristine phosphorene. Alternative strategies based on this observation may be developed in the future.

Conflicts of interest

The authors declare no conflicts of interest.

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References

- 1 L. Liu and A. Corma, *Chem. Rev.*, 2018, **118**, 4981–5079.
- 2 Y. Du, H. Sheng, D. Astruc and M. Zhu, *Chem. Rev.*, 2020, **120**, 526–622.

- 3 I. E. Stephens, A. S. Bondarenko, U. Grønberg, J. Rossmeisl and I. Chorkendor, *Energy Environ. Sci.*, 2012, **5**, 6744–6762.
- 4 X.-F. Yang, A. Wang, B. Qiao, J. Li, J. Liu and T. Zhang, *Acc. Chem. Res.*, 2013, **46**, 1740–1748.
- 5 H. Zhang, G. Liu, L. Shi and J. Ye, *Adv. Energy Mater.*, 2018, **8**, 1701343.
- 6 A. Wang, J. Li and T. Zhang, *Nat. Rev. Chem.*, 2018, **2**, 65–81.
- 7 W. Liu, L. Zhang, W. Yan, X. Liu, X. Yang, S. Miao, W. Wang, A. Wang and T. Zhang, *Chem. Sci.*, 2016, **7**, 5758–5764.
- 8 P. Xie, T. Pu, A. Nie, S. Hwang, S. C. Purdy, W. Yu, D. Su, J. T. Miller and C. Wang, *ACS Catal.*, 2018, **8**, 4044–4048.
- 9 T. Sun, L. Xu, D. Wang and Y. Li, *Nano Res.*, 2019, 1–14.
- 10 J. Lin, A. Wang, B. Qiao, X. Liu, X. Yang, X. Wang, J. Liang, J. Li, J. Liu and T. Zhang, *J. Am. Chem. Soc.*, 2013, **135**, 15314–15317.
- 11 M. F. Stephanopoulos, *Acc. Chem. Res.*, 2013, **47**, 783–792.
- 12 T. R. Reina, S. Ivanova, M. Centeno and J. A. Odriozola, *Catal. Today*, 2015, **253**, 149–154.
- 13 D.-W. Jeong, V. Subramanian, J.-O. Shim, W.-J. Jang, Y.-C. Seo, H.-S. Roh, J. H. Gu and Y. T. Lim, *Catalysis Letters*, 2013, **143**, 438–444.
- 14 Y. Zheng, Y. Jiao, Y. Zhu, Q. Cai, A. Vasile, L. H. Li, Y. Han, Y. Chen and S.-Z. Qiao, *J. Am. Chem. Soc.*, 2017, **139**, 3336–3339.
- 15 R.-Q. Zhang, T.-H. Lee, B.-D. Yu, C. Stamp and A. Soon, *Phys. Chem. Chem. Phys.*, 2012, **14**, 16552–16557.
- 16 N. Cheng, S. Stambula, D. Wang, M. N. Banis, J. Liu, A. Riese, B. Xiao, R. Li, T.-S. Sham, L.-M. Liu, G. A. Botton and X. Sun, *Nat. Comm.*, 2016, **7**, 13638.
- 17 S. K. Sahoo, Y. Ye, S. Lee, J. Park, H. Lee, J. Lee and W. Han, *ACS Ener. Lett.*, 2018, **4**, 126–132.
- 18 X. Mao, L. Zhang, G. Kour, S. Zhou, S. Wang, C. Yan, Z. Zhu and A. Du, *ACS Appl. Mater. Interfa.*, 2019, **11**, 17410–17415.
- 19 X. Mao, G. Kour, C. Yan, Z. Zhu and A. Du, *J. Phys. Chem. C*, 2019, **123**, 3703–3710.
- 20 X. Mao, C. Ling, C. Tang, C. Yan, Z. Zhu and A. Du, *J. Catal.*, 2018, **367**, 206–211.
- 21 H. Liu, A. T. Neal, Z. Zhu, Z. Luo, X. Xu, X. Tománek and P. D. Ye, *ACS Nano*, 2014, **8**, 4033–4041.
- 22 A. C.- Gomez, *J. Phys. Chem. Lett.*, 2015, **6**, 4280–4291.
- 23 J. Qiao, X. Kong, Z.-X. Hu, F. Yang and W. Ji, *Nat. Comm.*, 2014, **5**, 4475.
- 24 H. Liu, Y. Du, Y. Deng and D. Y. Peide, *Chem. Soc. Rev.*, 2015, **44**, 2732–2743.
- 25 L. Li, Y. Yu, G. J. Ye, Q. Ge, X. Ou, H. Wu, D. Feng, X. H. Chen and Y. Zhang, *Nature nanotechnology*, 2014, **9**, 372.
- 26 H. Tian, B. Deng, M. L. Chin, X. Yan, H. Jiang, S. J. Han, V. Sun, Q. Xia, M. Dubey, F. Xia and H. Wang, *ACS Nano*, 2016, **10**, 10428–10435.
- 27 D. K. Sang, H. Wang, Z. Guo, N. Xie and H. Zhang, *Adv. Funct. Mater.*, 2019, **29**, 1903419.
- 28 Q. Li, Q. Zhou, L. Shi, Q. Chen and J. Wang, *J. Mater. Chem. A*, 2019, **7**, 4291–4312.
- 29 M. Akhtar, G. Anderson, R. Zhao, A. Alruqi, J. E. Mroczkowska, G. Sumanasekera and J. B. Jasinski, *npj 2D Mater. Appl.*, 2017, **1**, 1–13.



- 30 J. Zhang, H. Shin and W. Lu, *Chem. Comm.*, 2019, **55**, 2601–2604.
- 31 X. Ren, J. Zhou, X. Qi, X. Liu, Z. Huang, Z. Li, Y. Ge, S. Chander, D. Joice, S. Ponraj, S. Wang, J. Zhong and H. Zhang, *Adv. Funct. Mater.*, 2017, **7**, 1700396.
- 32 X.-X. Xue, S. Shen, X. Jiang, P. Sengdala, K. Chen and Y. Feng, *J. Phys. Chem. Lett.*, 2019, **10**, 3440–3446.
- 33 G. Kresse and J. Hafner, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1994, **49**, 14251.
- 34 J. P. Perdew, J. A. Chevary, S. H. Vosko, K. A. Jackson, M. R. Pederson, D. J. Singh and C. Fiolhais, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1992, **46**, 6671.
- 35 G. Kresse and D. Joubert, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1999, **59**, 1758.
- 36 A. Hashmi and J. Hong, *J. Phys. Chem. C*, 2015, **119**, 9198–9204.
- 37 S. Grimme, J. Antony, S. Ehrlich and H. Krieg, *J. Chem. Phys.*, 2010, **132**, 154104.
- 38 K. Mathew, R. Sundararaman, K. Letchworth-Weaver, T. Arias and R. G. Hennig, *J. Chem. Phys.*, 2014, **140**, 084106.
- 39 R. Dronskowski and P. E. Blöchl, *J. Phys. Chem. C*, 1993, **97**, 8617–8624.
- 40 F. Banhart, J. Kotakoski and A. V. Krasheninnikov, *ACS Nano*, 2010, **5**, 26–41.
- 41 Y. Singh, S. Back and Y. Jung, *Phys. Chem. Chem. Phys.*, 2018, **20**, 21095–21104.
- 42 B. Garlyyev, J. Fichtner, O. Piqué, O. Schneider, A. S. Bandarenka and F. Calle-Vallejo, *Chem. Sci.*, 2019, **10**, 8060–8075.
- 43 W.-L. Li, C. Ertural, D. Bogdanovski, J. Li and R. Dronskowski, *Inorg. Chem.*, 2018, **57**, 12999–13008.
- 44 F. Calle-Vallejo, A. Krabbe and J. M. García-Lastra, *Chem. Sci.*, 2017, **8**, 124–130.
- 45 J. K. Nørskov, J. Rossmeisl, A. Logadottir, L. Lindqvist, J. R. Kitchin, T. Bligaard and H. Jonsson, *J. Phys. Chem. B*, 2004, **108**, 17886–17892.
- 46 I. C. Man, H.-Y. Su, F. Calle-Vallejo, H. A. Hansen, J. I. Martinez, N. G. Inoglu, J. Kitchin, T. F. Jaramillo, J. K. Nørskov and J. Rossmeisl, *ChemCatChem*, 2011, **3**, 1159–1165.
- 47 S. C. Perry, D. Pangotra, L. Vieira, L.-I. Csepei, V. Sieber, L. Wang, C. P. de León and F. C. Walsh, *Nat. Rev. Chem.*, 2019, **3**, 442–458.
- 48 A. Kulkarni, S. Siahrostami, A. Patel and J. K. Nørskov, *Chem. Rev.*, 2018, **118**, 2302–2312.
- 49 A. Mahata and B. Pathak, *Nanoscale*, 2017, **9**, 9537–9547.
- 50 S. Siahrostami, A. Verdaguier-Casadevall, M. Karamad, D. Deiana, P. Malacrida, B. Wickman, M. Escudero-Escribano, E. Paoli, R. Frydendal, T. W. Hansen, *et al.*, *Nat. Mater.*, 2013, **12**, 1137–1143.
- 51 Y. Abghoui and E. Skúlason, *J. Phys. Chem. C*, 2017, **121**, 24036–24045.
- 52 J. K. Nørskov, T. Bligaard, A. Logadottir, J. R. Kitchin, J. G. Chen, S. Pandelov and U. Stimming, *J. Elec. Chem. Soc.*, 2005, **152**, 23–26.
- 53 A. Ziletti, A. Carvalho, D. K. Campbell, D. F. Coker and A. C. Neto, *Phys. Rev. Lett.*, 2015, **114**, 046801.
- 54 G. Wang, W. J. Slough, R. Pandey and S. P. Karna, *2D Materials*, 2016, **3**, 025011.
- 55 Q. Zhou, Q. Chen, Y. Tong and J. Wang, *Angew. Chem. Int. Ed.*, 2016, **55**, 11437–11441.

