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## Highly microporous SbPO<sub>4</sub>/BC<sub>x</sub> hybrid anodes for sodium-ion batteries†

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The current anode materials greatly restrict the electrochemical performance of sodium-ion batteries. Herein, we propose a highly microporous SbPO<sub>4</sub>/BC<sub>x</sub> hybrid anode for sodium-ion batteries, exhibiting a high initial reversible capacity of 871 mA h g<sup>-1</sup> at 50 mA g<sup>-1</sup>, a good rate capability of around 300 mA h g<sup>-1</sup> even at 5 A g<sup>-1</sup> as well as an excellent cycling stability of 500 cycles. The excellent rate capability and cyclability with high capacity are probably due to the novel BC<sub>x</sub> structure and stable PO<sub>4</sub><sup>3-</sup> anions. The abundant micropores serve as reservoirs for storing the sodium ions and shorten the diffusion distance. The high surface area contributes to ample contact area between the electrode and electrolyte, thus achieving a rapid charge-transfer reaction. XPS analysis reveals that the BC<sub>x</sub> matrix consists of three B/C structures of BC<sub>3</sub>, BC<sub>2</sub>O, and BCO<sub>2</sub>, and contains around 12.93 at% substitutional boron. Since valence band holes are created by the B/C structures, more sodium ions would be captured easily, which motivates more sodium ions to intercalate electrochemically. Additionally, both the robust BC<sub>x</sub> matrix and stable PO<sub>4</sub><sup>3-</sup> anions as buffers could accommodate the volumetric expansion during the sodium ion insertion, thus optimizing the cycling performance. The strong attachment between SbPO<sub>4</sub> and the BC<sub>x</sub> matrix would benefit mutual charge transfer between them and keep the integrity of the electrode during the sodiation/desodiation processes, which are favorable for sodium-ion transport and play a crucial role in enhancing the rate performance. Accordingly, the SbPO<sub>4</sub>/BC<sub>x</sub> composite is expected to become a promising anode for advanced SIBs.

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### 1. Introduction

With regard to large-scale energy storage systems, sodium-ion batteries (SIBs) have gained extensive attention as potential alternatives to lithium-ion batteries (LIBs) due to the natural abundance and low cost of sodium and similar principles to LIBs.<sup>1–5</sup> Currently, the largest challenge of SIBs for practical applications is exploiting suitable electrode materials that can absorb/release sodium ions with enough reversible capacity and diffusion kinetics.<sup>6,7</sup> SIBs are not as attractive as LIBs, mostly because of the lack of anode materials with comparable performance to graphite anodes in LIBs. Among the current anode materials for SIBs, an alloy-type Sb anode is a promising candidate to enable high capacity (660 mA h g<sup>-1</sup>, Na<sub>3</sub>Sb),<sup>1,8</sup> high electronic conductivity, and considerably safe operating

potential (0.6 V vs. Na<sup>+</sup>/Na).<sup>9,10</sup> However, metallic Sb anodes are often subjected to large volume changes (390%) upon cycling,<sup>11–13</sup> which would easily lead to rapid capacity decay.

To alleviate the large volume changes upon cycling of the Sb anodes, carbon buffer was proposed and various composites, such as Sb/hard carbon,<sup>13–18</sup> Sb/carbon black,<sup>11</sup> Sb/carbon fibers,<sup>19</sup> Sb/carbon nanotubes,<sup>20</sup> and Sb/graphene,<sup>7,21–24</sup> have been reported. These carbon matrices, especially hard carbon, also contribute to the sodium storage capacity of about 300 mA h g<sup>-1</sup>,<sup>25–27</sup> yet more than 0.5 times capacity of the carbon matrix was obtained at a low plateau (0.1 V vs. Na/Na<sup>+</sup>), further triggering security issues during the fast charging processes.<sup>21,27,28</sup> Additionally, the carbon matrix suffers from slow ion diffusion kinetics.<sup>29,30</sup> These deficiencies of metallic Sb anodes and the carbon matrix thus severely hamper their practical applications in SIBs, which urges the exploitation of novel and robust anode materials for SIBs. Recently, layered-structure SbPO<sub>4</sub> was explored, for the first time, by Qian *et al.*,<sup>7</sup> exhibiting hopeful potential as an anode material for SIBs, because bulky and stable PO<sub>4</sub><sup>3-</sup> anions can buffer the volume change upon cycling and facilitate the cycling stability. Also, Na<sub>3</sub>PO<sub>4</sub> generated by the discharge reaction, as an ionic conductor, may reduce the diffusion barrier of sodium ions and promote the

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reaction kinetics. Accordingly, if the conversion of Sb to SbPO<sub>4</sub> over the Sb/C anode material was achieved, it would greatly improve the electrochemical performances of pristine metallic Sb anodes. On the other hand, the heteroatom doping and structure engineering of carbon is beneficial to increasing the sodium storage capacity of Sb/C anodes at the expense of the initial Coulombic efficiency and cycling stability.<sup>31–34</sup> Most typically, some documents<sup>35–38</sup> were aimed at the novel graphite-like BC<sub>3</sub> structure as an anode material for alkali metal ion batteries. Many simulation results<sup>35,39</sup> suggest that sodium could effectively intercalate this structure with the maximum sodium concentration, and the much lower potential barrier of BC<sub>3</sub> promotes the sodium transportation, which is helpful in increasing the diffusion rates of sodium atoms inside the BC<sub>3</sub> structure. Thus, these excellent features of BC<sub>3</sub> encourage further experimental studies that the BC<sub>3</sub> structure was used to prepare the volume buffer of the Sb anodes.

Herein, we propose the preparation of SbPO<sub>4</sub>/BC<sub>x</sub> hybrid anodes for sodium-ion batteries by a facile conversion of Sb to SbPO<sub>4</sub> through one-step annealing of a solid mixed powder of Sb/BC<sub>x</sub> and NH<sub>4</sub>H<sub>2</sub>PO<sub>4</sub>, in which the Sb/BC<sub>x</sub> composites were fabricated by pyrolyzing antimony acetate impregnated ammonium pentaborate/starch xerogels at 800 °C under an argon atmosphere. The SbPO<sub>4</sub>/BC<sub>x</sub> hybrid electrode could provide stable PO<sub>4</sub><sup>3–</sup> anions and sufficient void space from the BC<sub>x</sub> matrix, which can perfectly accommodate the volumetric expansion during the sodium ion insertion, and preserve the structural stability of anodes. As a result, the SbPO<sub>4</sub>/BC<sub>x</sub> composite as an anode of SIBs exhibits a high initial reversible capacity of 871 mA h g<sup>–1</sup> at 50 mA g<sup>–1</sup>, a good rate capability of around 300 mA h g<sup>–1</sup> even at 5 A g<sup>–1</sup> and an excellent cycling stability of 500 cycles. Hence, the SbPO<sub>4</sub>/BC<sub>x</sub> composite is expected to have promising application in future high-performance SIBs.

## 2. Material and methods

### 2.1. Material preparation

The Sb/BC<sub>x</sub> composite was first synthesized by pyrolyzing antimony acetate impregnated ammonium pentaborate/starch xerogels (Fig. 1). Briefly, 0.7 g of Sb(CH<sub>3</sub>COO)<sub>3</sub> and 0.15 g of NH<sub>4</sub>B<sub>5</sub>O<sub>8</sub> were dissolved in turn in 20 mL of deionized water.

Next, 0.3 g of soluble starch was added to form a white solution. The mixed solution was heated at 85 °C for 1 h under vigorous stirring to obtain a transparent hydrogel. After freeze-drying, the resultant xerogels were carbonized at 800 °C for 2 h under an argon atmosphere, forming the Sb/BC<sub>x</sub> composites (denoted as Sb/BC<sub>x</sub>). The Sb/C composites and Sb were prepared using a similar method, respectively, and used as the control sample. Afterwards, 0.5 g of the as-prepared Sb/BC<sub>x</sub> was fully mixed with 5 g of NaH<sub>2</sub>PO<sub>4</sub> powder through hand-grinding, and then the mixture was heated to 400 °C at a rate of 10 °C min<sup>–1</sup>, and annealed for 1 h in a H<sub>2</sub>/Ar (5:95) atmosphere. The calcined products were washed with deionized water and alcohol a few times, and dried at 60 °C (denoted as SbPO<sub>4</sub>/BC<sub>x</sub>).

### 2.2. Structural characterization

X-ray diffraction (XRD) of all the samples was performed on a Bruker AXS D8 with Cu-Kα as a radiation source (λ = 1.5406 Å). Raman spectra were obtained on inVia instrument (Renishaw) with a 532 nm Ar-ion laser as an excitation source. Their morphologies and microstructure were analysed by scanning electron microscopy (SEM, S4800, Hitachi) and transmission electron microscopy (TEM, FEI Tecnai G20). X-ray photoelectron spectroscopy (XPS) was determined on a Thermo spectrometer using Al-Kα radiation (Thermo Scientific, ESCALAN 250Xi). N<sub>2</sub> sorption isotherms were performed on a gas sorptometer at 77 K (Micrometrics ASAP 2020 analyser). Fourier transform infrared (FT-IR) spectra were recorded on a Nicolet 6700 FTIR spectrophotometer over the wavenumber range of 4000–500 cm<sup>–1</sup>.

### 2.3. Electrochemical measurements

The electrochemical performance of SbPO<sub>4</sub>/BC<sub>x</sub> as an anode material in SIBs was studied using CR2025-type coin cells. In each test, the working electrode was prepared by mixing 80 wt% active material, 10 wt% acetylene black, and 10 wt% sodium alginate to form a black slurry, which was then coated on a clean copper foil and dried at 60 °C overnight under vacuum conditions. The mass loading of active materials was about 1.0–2.0 mg cm<sup>–2</sup>. The cells were assembled in an Ar-filled glove box (MIKROUNA, O<sub>2</sub> < 1 ppm, H<sub>2</sub>O < 1 ppm) with sodium foil as the reference and counter electrodes, 1 M NaClO<sub>6</sub> solution in EC/DEC (1:1 by volume) solvent with 5% fluoroethylene

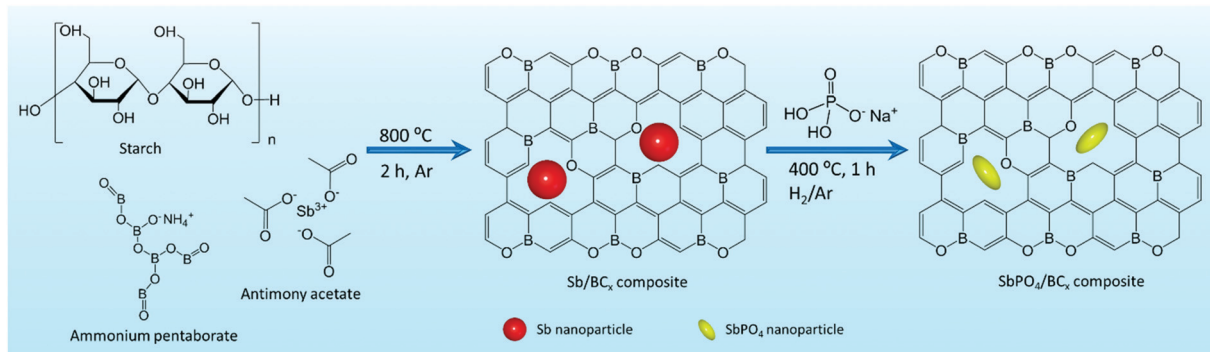


Fig. 1 Schematic illustration of the preparation process for the SbPO<sub>4</sub>/BC<sub>x</sub> composite.



carbonate as the electrolyte, and the GF/D glass microfiber filter (Whatman) as the separator. Galvanostatic discharge/charge tests were performed on a battery test system (NEWARE CT4008) with a voltage range from 0.01 to 3.0 V. All the capacity values were determined based on the total weight of the active material in the working electrodes. Cyclic voltammetry (CV) was carried out with coin cells at a scan rate of  $0.1 \text{ mV s}^{-1}$  using a CHI 760E electrochemical workstation. Electrochemical impedance spectroscopy (EIS) of coin cells was performed using an SP200 electrochemical workstation in the frequency range of  $10^{-2}$ – $10^5$  Hz by applying an AC voltage of 1 mV amplitude. The Nyquist curves are fitted by Zview software. All the tests were performed at room temperature.

### 3. Results and discussion

The  $\text{SbPO}_4/\text{BC}_x$  composite was prepared by a solid-phase annealing process (Fig. 1). The morphological characteristics of the precursor  $\text{Sb}/\text{BC}_x$  after pyrolysis at  $800^\circ\text{C}$  under an argon atmosphere are shown in Fig. 2. The spherical Sb nanoparticles with a mean size of 22 nm are uniformly anchored within the

$\text{BC}_x$  matrix as presented by TEM images (Fig. 2a and b). The HRTEM image in Fig. 2c reveals clear lattice fringes with a  $d$ -spacing of around 0.31 nm corresponding to the (021) lattice planes of hexagonal Sb. For comparison, the TEM images of the  $\text{Sb}/\text{C}$  composite are shown in Fig. 2d and e, where the Sb nanoparticles with a mean size of 24 nm are unevenly embedded within the carbon matrix, and some cavities can be clearly seen since some Sb nanoparticles are shed from the carbon matrix, in contrast to the Sb nanoparticles in the  $\text{Sb}/\text{BC}_x$  composite that are attached closely to the  $\text{BC}_x$  matrix. Besides, the HRTEM image of Fig. 2f presents lattice fringes with a  $d$ -spacing of around 0.31 nm corresponding to the (021) lattice plane for hexagonal Sb, which is consistent with the results from the HRTEM image of the  $\text{Sb}/\text{BC}_x$  composite. The distribution of Sb in the  $\text{BC}_x$  matrix was confirmed by a scanning transmission electron microscopy (STEM) image, where Sb is identified as white nanodots. The TEM-EDS elemental mappings of Sb, C, B and O correspond to the area shown in Fig. 2g, and the four elements are uniformly distributed over the entire sample. The presence of the O element may be due to the highly porous structure of the  $\text{Sb}/\text{BC}_x$  composite, which can easily absorb oxygen from air on the surface of the material.



Fig. 2 TEM images (a and b) and HRTEM images (c) of the  $\text{Sb}/\text{BC}_x$  composite; TEM images (d and e) and HRTEM images (f) of the  $\text{Sb}/\text{C}$  composite; (g) scanning TEM image with its corresponding EDS mappings of the Sb, C, B and O elements for the  $\text{Sb}/\text{BC}_x$  composite.

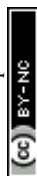




Fig. 3 SEM image (a), TEM and HRTEM images (b), and scanning TEM image with the corresponding EDS mappings of the Sb, P, O, C and B elements (c) of the  $\text{SbPO}_4/\text{BC}_x$  composite; (d) XRD patterns, (e) Raman spectra and (f) FT-IR spectra of the  $\text{SbPO}_4/\text{BC}_x$ ,  $\text{Sb}/\text{BC}_x$  and  $\text{Sb}/\text{C}$  samples. The two insets present the SAED pattern and the FFT pattern of the HRTEM image.

The characteristics of the  $\text{SbPO}_4/\text{BC}_x$  composite formed by one-step annealing of the solid powder of  $\text{Sb}/\text{BC}_x$  and  $\text{NH}_4\text{H}_2\text{PO}_4$  are shown in Fig. 3. The SEM image in Fig. 3a reveals that the  $\text{SbPO}_4$  nanoparticles are randomly dispersed on the  $\text{BC}_x$  matrix. The TEM image in Fig. 3b also indicates that there are numerous short rod-like nanoparticles with a diameter of about 50 nm and a length of up to 100 nm. Also, there are a small number of spindles in the  $\text{SbPO}_4/\text{BC}_x$  composite. The selected area electron diffraction (SAED) and the corresponding fast Fourier transform (FFT) patterns (the inset of Fig. 3b) can be indexed to monoclinic  $\text{SbPO}_4$ ,<sup>40,41</sup> indicating the presence of a pure phase of  $\text{SbPO}_4$  in the material. The lattice-resolved HRTEM image of a single nanoparticle in Fig. 3b shows clear lattice fringes with a  $d$ -spacing of 0.41 nm, assigned to the (020) planes of  $\text{SbPO}_4$ .<sup>7</sup> In the HRTEM image of the  $\text{BC}_x$  matrix, no obvious long-range-order structure exists, suggesting the amorphous nature of the material. A large number of defects and pores are distributed within the  $\text{BC}_x$  matrix. The STEM image of the  $\text{SbPO}_4/\text{BC}_x$  composite and the corresponding TEM-EDS element mappings display the good distribution of the Sb, P, O, C and B elements in the material (Fig. 3c). All the

above results confirm that the  $\text{SbPO}_4$  nanoparticles adhere well to the  $\text{BC}_x$  matrix. The strong attachment would benefit the charge transfer between  $\text{SbPO}_4$  and the  $\text{BC}_x$  matrix and retain the integrity of the electrode upon cycling, thereby improving the cycling stability and rate capability.<sup>14,15,25</sup> The XRD patterns in Fig. 3d confirm the phase constituents of the material during each stage of the preparation of the  $\text{SbPO}_4/\text{BC}_x$  composite. The precursor  $\text{Sb}/\text{BC}_x$  displays two phases consisting of Sb (JCPDS Card No. 85-1322) and  $\text{Sb}_2\text{O}_3$  (JCPDS Card No. 43-1071). Compared with the  $\text{Sb}/\text{C}$  sample, a majority of the  $\text{Sb}_2\text{O}_3$  phase in  $\text{Sb}/\text{BC}_x$  is reduced to form metallic Sb after doping boron into the carbon matrix, since boron atoms can boost the carbon graphitization process, which largely accelerates the conversion from  $\text{Sb}_2\text{O}_3$  to Sb. The resultant  $\text{SbPO}_4/\text{BC}_x$  composite displays multiphase components assigned to  $\text{SbPO}_4$  (JCPDS Card No. 78-1791) and Sb (JCPDS Card No. 85-1322). In addition to the six diffraction peaks of  $\text{SbPO}_4$  at  $21.5^\circ$ ,  $24.3^\circ$ ,  $26.1^\circ$ ,  $29.6^\circ$ ,  $37.5^\circ$  and  $43.7^\circ$ , it shows the diffraction peaks of Sb at  $28.6^\circ$ ,  $40.1^\circ$ ,  $41.8^\circ$ , and  $51.6^\circ$ . Additionally, no impurity peaks were detected, indicating that the  $\text{Sb}_2\text{O}_3$  phase was completely converted into the  $\text{SbPO}_4$  phase. The phase conversion of Sb to  $\text{SbPO}_4$  can thus



be easily achieved on the  $BC_x$  matrix through one-step annealing of the solid mixed powder of  $Sb/BC_x$  and  $NH_4H_2PO_4$ . The ratios of Sb species were obtained by quantitative analysis through the Rietveld refinement of the XRD pattern (Fig. S1, ESI<sup>†</sup>), and the results are summarized in Table S1 (ESI<sup>†</sup>). The ratio of Sb to  $SbPO_4$  in  $SbPO_4/BC_x$  and the ratio of Sb to  $Sb_2O_3$  in  $Sb/BC_x$  and  $Sb/C$  were about 11.4/88.6, 78.1/21.9, and 28.4/71.6, respectively. The proportion of the Sb phase in  $SbPO_4/BC_x$  is much lower than that of the precursor  $Sb/BC_x$ , hinting that metallic Sb is also partially converted to  $SbPO_4$ .

To examine the crystallinity of the  $BC_x$  matrix, Raman spectra of the  $SbPO_4/BC_x$ ,  $Sb/BC_x$  and  $Sb/C$  samples are recorded (Fig. 3e). Except for the peaks around  $456\text{ cm}^{-1}$ , corresponding to typical metallic Sb,<sup>25</sup> all the Raman spectra exhibit two prominent peaks at  $1340$  and  $1580\text{ cm}^{-1}$  assigned to the disordered carbon with  $sp^3$  hybridization (D-band) and graphitic carbon with  $sp^2$  hybridization (G-band), respectively.<sup>29,36</sup> The integrated intensity ratio of the D and G bands ( $I_D/I_G$ ) for the  $Sb/BC_x$  composite is 1.08, which is lower than that (2.21) of  $Sb/C$ , revealing that the  $BC_x$  matrix has a higher graphitization degree.<sup>23,42</sup> In the case of  $SbPO_4/BC_x$ , the value of  $I_D/I_G$  ratio is 1.11, nearly equal to that of  $Sb/BC_x$ , suggesting that the phase conversion of Sb to  $SbPO_4$  almost does not affect the  $BC_x$  structure. FT-IR characterization was also carried out on these samples to elaborate the chemical bonds (Fig. 3f). In all the samples, the broad peak located at  $\sim 3445\text{ cm}^{-1}$  was assigned to the O–H stretching vibration, and the peak centered at  $1612\text{ cm}^{-1}$  was ascribed to the C=C stretching vibrations. In the  $SbPO_4/BC_x$  sample, a prominent feature assigned to P–O stretching vibrations

at  $\sim 1054$  and  $\sim 987\text{ cm}^{-1}$  is visible, probably assigned to  $SbPO_4$ .<sup>40</sup> Except for the  $Sb/C$  sample, two extra peaks detected at  $1190$  and  $1410\text{ cm}^{-1}$  in the  $SbPO_4/BC_x$  and  $Sb/BC_x$  samples were attributed to B–C and B–O stretching vibrations,<sup>43,44</sup> respectively, indicating the successful formation of the  $BC_x$  structure by the substitution of boron into the carbon lattice.<sup>45</sup>

X-ray photoelectron spectroscopy (XPS) was conducted to reveal the composition and the chemical bonding states in the  $SbPO_4/BC_x$  composite. The signals of Sb, P, O, C and B could be easily detected in the survey spectrum (Fig. 4a). The high-resolution spectra of these elements confirm the existence of the  $SbPO_4$  and  $BC_x$  structure (Fig. 4b–e). The high-resolution Sb 3d peaks for  $SbPO_4/BC_x$  are fitted into four peaks (Fig. 4b), which are assigned to Sb  $3d_{5/2}$  ( $530.3\text{ eV}$ ) and Sb  $3d_{3/2}$  ( $539.7\text{ eV}$ ) of  $Sb^{3+}$ , and Sb  $3d_{5/2}$  ( $528.4\text{ eV}$ ) and Sb  $3d_{3/2}$  ( $537.6\text{ eV}$ ) of  $Sb^0$ , respectively.<sup>6,7</sup> Note that there are two O 1s peaks located at  $532.9\text{ eV}$  and  $531.7\text{ eV}$ , which are assigned to the P–O bonding in  $P_2O_5$  and  $PO_4^{3-}$ , respectively.<sup>14</sup> For comparison, the high-resolution Sb 3d peaks of  $Sb/BC_x$  and  $Sb/C$  are also shown (Fig. S2, ESI<sup>†</sup>). The high-resolution P 2p peaks centered at  $134.6\text{ eV}$  for  $SbPO_4/BC_x$  are fitted into two peaks (Fig. 4c), which are assigned to P–O (phosphate,  $134.3\text{ eV}$ ) and P–O (phosphorus oxide,  $135.7\text{ eV}$ ), respectively.<sup>46,47</sup> These XPS results of Sb 3d and P 2p further confirm that the  $SbPO_4/BC_x$  composite consists of  $SbPO_4$  and Sb phases, in good accord with the XRD results (Fig. 3d). The high-resolution B 1s peaks for the  $BC_x$  matrix are fitted into three peaks (Fig. 4d and Fig. S2, ESI<sup>†</sup>), corresponding to three B/C structures<sup>36,43</sup> of  $BC_3$  ( $190.2\text{ eV}$ ),  $BC_2O$  ( $191.5\text{ eV}$ ), and  $BCO_2$  ( $193.2\text{ eV}$ ), and the other boron

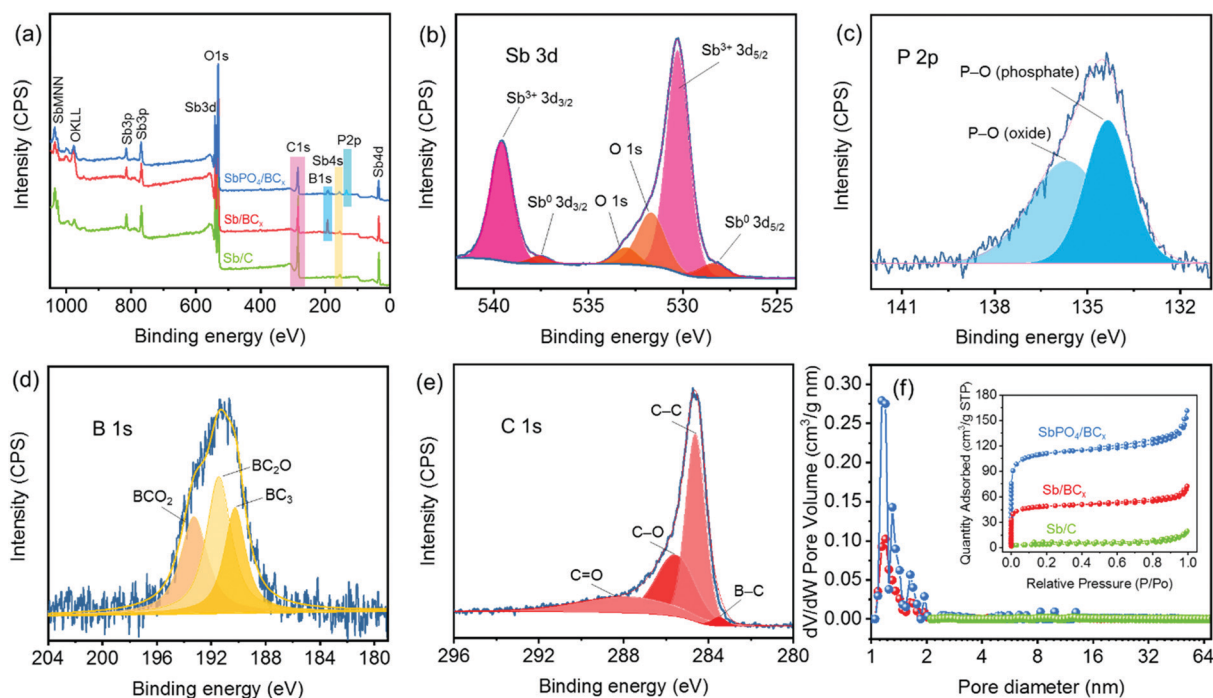


Fig. 4 (a) XPS survey of  $SbPO_4/BC_x$ ,  $Sb/BC_x$  and  $Sb/C$ , and the fitted Sb 3d signal (b), P 2p signal (c), B 1s signal (d), and C 1s signal (e) for the  $SbPO_4/BC_x$  composite. (f) NLDFT adsorption pore size distribution plots (the inset presents  $N_2$  adsorption–desorption isotherm curves) of the  $SbPO_4/BC_x$ ,  $Sb/BC_x$  and  $Sb/C$  samples.





Fig. 3e presents the Nyquist plots of the  $\text{SbPO}_4/\text{BC}_x$ ,  $\text{Sb}/\text{BC}_x$ ,  $\text{Sb}/\text{C}$  and  $\text{Sb}$  electrodes; all the spectra consist of semicircles in the medium–high frequency region and sloping straight lines in the low-frequency region, which are related to the charge-transfer resistance ( $R_{ct}$ ) and solid-state diffusion of sodium in the electrode material, respectively.<sup>7,20</sup> As shown in Table S3 (ESI<sup>†</sup>), the  $\text{SbPO}_4/\text{BC}_x$  electrode exhibits the lowest charge-transfer resistance ( $R_{ct} = 44.43 \Omega$ ), which indicates its high electrical conductivity and enhanced reaction kinetics upon cycling. The results obtained from the 4-point probe method further confirm that the electronic conductivity of  $\text{SbPO}_4/\text{BC}_x$  is much larger than those of  $\text{Sb}/\text{BC}_x$  and  $\text{Sb}/\text{C}$  (Table S3, ESI<sup>†</sup>). The relationship between  $Z_{re}$  and  $\omega^{-1/2}$  in Fig. 3f represents the Warburg impedance related to sodium-ion diffusion. The  $\text{SbPO}_4/\text{BC}_x$  electrode displays the lowest slope ( $\sigma = 9.18$ ) among the three electrodes (Table S3, ESI<sup>†</sup>), implying facile sodium-ion transportation. The EIS result is consistent with its excellent cycling and rate performance, hinting the reinforced electron and ion transport. These results thus suggest that the robust B/C structure and stable  $\text{PO}_4^{3-}$  anions are favorable for sodium-ion transport and play an important role in enhancing the rate performance (Fig. 5).

A long-term cycling at a high current rate of  $5 \text{ A g}^{-1}$  is achieved for the  $\text{SbPO}_4/\text{BC}_x$  electrode (Fig. 6a). After a few activation cycles under  $0.2 \text{ A g}^{-1}$ , the capacity remains at  $300 \text{ mA h g}^{-1}$ . In the following cycles, the capacity decreases to  $127 \text{ mA h g}^{-1}$  after 150 cycles,  $118 \text{ mA h g}^{-1}$  after 300 cycles, and  $112 \text{ mA h g}^{-1}$  after 500 cycles. Thus, the  $\text{SbPO}_4/\text{BC}_x$  electrode at a high current rate exhibits the low capacity retention of 37.3% as well as good cycling stability. The large

initial capacity fading of  $\text{SbPO}_4/\text{BC}_x$  is probably attributed to the restricted buffering ability to adjust the volume expansion, which can be proved by observing the morphologies of electrodes before and after cycling (Fig. S3, ESI<sup>†</sup>). After 500 cycles, a small number of cracks are observed on the surface of the  $\text{SbPO}_4/\text{BC}_x$  electrode, which make it hard to retain the integrity of the electrode upon cycling. The TEM images in Fig. 6b show that the  $\text{BC}_x$  matrix still retains its primary structure after 500 cycles, accounting for the good cycling stability of the  $\text{SbPO}_4/\text{BC}_x$  electrode at a high current rate.

The excellent rate capability and cyclability with high capacity could be illustrated by the robust  $\text{BC}_x$  structure and stable  $\text{PO}_4^{3-}$  anions (Fig. 6c). The amorphous  $\text{BC}_x$  structure with a large  $d$ -spacing of  $\sim 0.41 \text{ nm}$  can intercalate more sodium ions. The abundant micropores serve as reservoirs for storing the sodium ions and shorten the diffusion distance. The high  $S_{\text{BET}}$  value leads to the large contact area between the electrode and electrolyte, thus exhibiting the rapid charge-transfer reaction. Additionally, since valence band holes are created by boron doping, more sodium ions would be captured easily in the  $\text{BC}_x$  structure, which motivates more sodium ions to intercalate electrochemically. Both the robust  $\text{BC}_x$  matrix and stable  $\text{PO}_4^{3-}$  anions as buffers could alleviate the volume change upon cycling, optimizing the cycling performance. The strong attachment between  $\text{SbPO}_4$  and  $\text{BC}_x$  matrix would benefit mutual charge transfer between them, and keep the integrity of the electrode during the sodiation/desodiation processes. All of these merits make the  $\text{SbPO}_4/\text{BC}_x$  composite promising as an anode material for SIBs.



Fig. 5 (a) CV curves of the  $\text{SbPO}_4/\text{BC}_x$  electrode at a scan rate of  $0.1 \text{ mV s}^{-1}$ . (b) Galvanostatic charging/discharging curves of the  $\text{SbPO}_4/\text{BC}_x$  electrode at a current rate of  $0.2 \text{ A g}^{-1}$  over  $3.0-0.01 \text{ V}$ . (c) Cycle performance of the  $\text{SbPO}_4/\text{BC}_x$ ,  $\text{Sb}/\text{BC}_x$ ,  $\text{Sb}/\text{C}$  and  $\text{Sb}$  electrodes at a current rate of  $0.2 \text{ A g}^{-1}$  over  $3.0-0.01 \text{ V}$ . (d) Rate performance of the  $\text{SbPO}_4/\text{BC}_x$ ,  $\text{Sb}/\text{BC}_x$ ,  $\text{Sb}/\text{C}$  and  $\text{Sb}$  electrodes at various current rates from  $0.1$  to  $5.0 \text{ A g}^{-1}$ . Nyquist impedance plots and the magnified area (inset) in the range of  $0-225 \text{ Ohm}$  (e) and the corresponding relationship between the real part of the impedance ( $Z_{re}$ ) and  $\omega^{-1/2}$  (f) of the  $\text{SbPO}_4/\text{BC}_x$ ,  $\text{Sb}/\text{BC}_x$ ,  $\text{Sb}/\text{C}$  and  $\text{Sb}$  electrodes.





Fig. 6 (a) Cycling performance of the  $\text{SbPO}_4/\text{BC}_x$  electrode at  $5 \text{ A g}^{-1}$  after three cycles at  $0.2 \text{ A g}^{-1}$ . (b) TEM images of the  $\text{SbPO}_4/\text{BC}_x$  electrode after 500 cycles. (c) Proposed mechanism of the  $\text{SbPO}_4/\text{BC}_x$  composite during the sodium storage processes.

## 4. Conclusions

In summary, we propose a  $\text{SbPO}_4/\text{BC}_x$  hybrid anode for sodium-ion batteries by a facile conversion of Sb to  $\text{SbPO}_4$  through one-step annealing of a solid mixed powder of  $\text{Sb}/\text{BC}_x$  and  $\text{NH}_4\text{H}_2\text{PO}_4$ , where the  $\text{Sb}/\text{BC}_x$  composites are prepared by pyrolyzing antimony acetate impregnated ammonium pentaborate/starch xerogels under an Ar atmosphere at  $800^\circ\text{C}$ . When used as the anode of SIBs, the  $\text{SbPO}_4/\text{BC}_x$  composite exhibits a high initial reversible capacity of  $871 \text{ mA h g}^{-1}$  at  $50 \text{ mA g}^{-1}$ , a good rate capability of about  $300 \text{ mA h g}^{-1}$  even at  $5 \text{ A g}^{-1}$  and an excellent cycling stability of 500 cycles. These results confirm that the excellent cycling performance of the  $\text{SbPO}_4/\text{BC}_x$  electrode is ascribed to the prominent advantages of a highly microporous structure, the buffer of volume change (stable  $\text{PO}_4^{3-}$  anions), the lower potential barrier of  $\text{BC}_3$ , and abundant edge defects ( $\text{BC}_2\text{O}$ ,  $\text{BCO}_2$ ). The novel B/C structure and stable  $\text{PO}_4^{3-}$  anions are favorable for sodium-ion transport and play an important role in enhancing the rate performance. It is believed that this easy method of combining  $\text{SbPO}_4$  with a  $\text{BC}_x$  matrix might be extended to other anode materials for reinforced electrochemical performance of SIBs.

## Conflicts of interest

There are no conflicts to declare.

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