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# Poly(lactic acid) stereocomplex microspheres as thermally tolerant optical resonators

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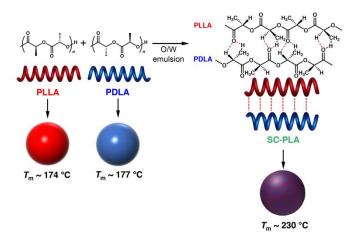
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Thermally tolerant polymer optical resonators are fabricated from stereocomplex of poly(L-lactic acid) and poly(D-lactic acid) through oil-in-water miniemulsion method. The thermal stability of the microspheres of the stereocomplex poly(lactic acid) (SC-PLA) is superior to those of the homochiral poly(lactic acid) (HC-PLA). As a result of the high thermal stability, optical resonator properties of the SC-PLA microspheres preserve at elevated temperature up to 230 °C, which is higher by 70 °C than that of microspheres formed from HC-PLA.

Poly(lactic acid) (PLA) is a synthetic polyester made from lactic acid. PLA is widely used for biomedical applications, PLA is widely used for biomedical applications, PLA has three types of stereoisomers; poly(L-lactide acid) (PLLA), poly(D-Lactide acid) (PDLA), and their atactic polymer, poly(DL-lactic acid)(PDLLA). The homopolymers of PLLA and PDLA form semicrystalline aggregates with a melting temperature at around 170 °C, while the atactic PDLLA copolymer forms an amorphous aggregate with a lower melting temperature. Place is a synthetic polymer and polymer and polymer aggregate with a lower melting temperature.

One of the drawbacks of PLA is that the thermal stability of PLA is not so high due to its low crystallinity, which limits the use of PLA in optical and electronic applications  $^{14}$ . Some techniques have been applied to enhance the thermal stability of PLA by adding a nucleating agent,  $^{15-18}$  physical modification through fibre reinforcement,  $^{19,20}$  the addition of inorganic particles  $^{21}$ , chemical modification,  $^{22}$  and blending with the material with high  $T_{\rm g}$  and high heat resistance,  $^{23,24}$  In particular, stereocomplexation by mixing PLLA and PDLA have been received much attention due to its enhanced thermal and mechanical properties.  $^{25,}$   $^{26}$  A mixture of PLLA and PDLA,

prepared from its molten state or solution state, induces the formation of stereocomplex PLA (SC-PLA) driven by the intensive intermolecular hydrogen (H)-bonding and dipole-dipole interactions between PLLA and PDLA (**Fig. 1**). $^{27-29}$  Compared with homochiral PLA (HC-PLA), SC-PLA has a melting temperature ( $T_{\rm m}$ ) of around 230 °C, which is higher than that of the pure HC-PLA (~170 °C), since the intermolecular H-bonding interaction between PLLA and PDLA chains increases the rigidity of the PLA chains. $^{30-35}$ 



**Fig. 1.** Schematic illustration of fabrication of microspheres from HC-and SC-PLA by O/W emulsion method.

Optical resonators confine light in a small volume that interferes to show up sharp optical signals.<sup>36</sup> Typical optical resonators are Fabry-Pérot (F-P) resonators and whispering gallery mode (WGM) resonators.<sup>37</sup> For F-P resonators, light is confined between counter mirrors or crystalline facets.<sup>38</sup> On the other hand, WGM resonators confine light circularly by total internal reflection (TIR) at the interface between the inner and outer media with different refractive indices. Typical WGM resonators have shapes of sphere, ring, polyhedron etc.<sup>39</sup> WGM is highly sensitive to the surface morphology, because a rough surface tends to scatter light, which reduce the light confinement efficiency in the resonator.<sup>40</sup> Therefore, highly smooth surface is one of important factors for the high quality factor (*Q*) microresonators. The sharp PL peaks from the

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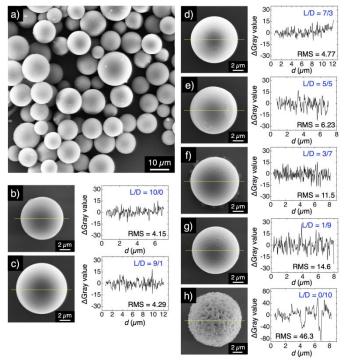
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resonators are utilized for highly sensitive chemical and biosensing applications by monitoring the peak shift. 41,42

In this work, we investigate optical resonator properties of the microspheres formed from PLAs. The whispering gallery mode (WGM) microresonators are fabricated from a blend of PLLA and PDLA through oil-in-water (O/W) miniemulsion method. We expect that the microresonators formed from SC-PLA has better thermal stability than the microresonators formed from pure HC-PLA. We find that the WGM resonators from SC-PLA are much more stable even at 230 °C in comparison with those from HC-PLA (~170 °C).

Microspheres of HC-PLA composed of PLLA (the number averaged molecular weight,  $M_{\rm n}$  = 40 kg mol<sup>-1</sup>) or PDLA ( $M_{\rm n}$  = 90 kg mol<sup>-1</sup>) were prepared by O/W miniemulsion method. Similarly, microspheres of SC-PLA composed of a blend of PLLA and PDLA were fabricated with the mixing ratio of PLLA to PDLA (L/D) of 9/1, 7/3, 5/5, 3/7, and 1/9. The miniemulsion method is chosen because of the efficient formation of SC-PLA with high reaction rate compared to other methods such as solution blending, supercritical fluid, and melted blending. <sup>32,43</sup> During the emulsification and subsequent solvent evaporation processes, PLLA and PDLA form stereocomplex.

Fig. 2 shows scanning electron microscopy (SEM) images of the resultant microspheres formed from HC- and SC-PLA. All the microspheres have high sphericity but have different surface morphology. The microspheres of HC-PLA with L/D of 10/0 containing only PLLA (Fig. 2a and b) and SC-PLA with L/D of 9/1 and 7/3 (Fig. 2c and d, respectively) have rather smooth surface with the values of the root-mean-square (RMS) roughness between 4-5 (in details of the calculation of the RMS roughness, see the Supporting Information). The surface of the microsphere becomes rough with the increase of the content of PDLA with L/D of 5/5, 3/7 and 1/9 with the value of the RMS roughness of 6.23, 11.5, and 14.6, respectively (Fig. 2e-g). As for the microparticles made from only PDLA (L/D = 0/10), the surface morphology is quite rough with the RMS roughness value as large as 46.3 (Fig. 2h). The difference of the surface morphologies of the microspheres from PLLA and PDLA is possibly due to the difference in the molecular weights of these polymers, which causes the different crystallinity of the microspheres.44



**Fig. 2.** SEM micrographs of HC-PLA and SC-PLA microspheres prepared by O/W miniemulsion method with different L/D ratios of 10/0 (a, b) 9/1 (c), 7/3 (d), 5/5 (e), 3/7 (f), 1/9 (g), and 0/10 (h). The graphs on the right in (b–h) show profiles of the difference of the gray values versus distance (d) at the cross section of the microspheres shown as yellow-dotted lines in the corresponding micrographs.

We investigate the change in the morphology of the microsphere by heating. As shown in **Fig. S1**, the microspheres of HC-PLA (L/D = 10/0 and 0/10) melt when heated at  $200\,^{\circ}\text{C}$  for 2 s. However, microspheres of SC-PLA with the L/D ratio of 9/1, 7/3, 5/5, 3/7, and 1/9 keep their spherical morphologies after being heated at  $200\,^{\circ}\text{C}$ , indicating that the thermal stability of the PLA is remarkably improved by the SC formation. Even for a microsphere with L/D of 9/1, where the composition of the HC crystallites is greater than that of the SC crystallites, the microsphere preserves its spherical morphology upon heating at  $200\,^{\circ}\text{C}$ .

The thermal stability of HC- and SC-PLA microspheres was investigated in more detail by differential scanning calorimetry (DSC). **Fig. 3** shows the DSC thermograms (first heating) of the HC and SC crystallites with the different mixing ratios (L/D = 10/0, 9/1, 7/3, 5/5, 3/7, 1/9, 0/10). Several characteristic endo-/exothermic peaks were observed: Endothermic peaks at 60-70 °C due to the glass transition ( $T_g$ ), exothermic peaks at 90-110 °C due to the crystallization ( $T_c$ ), and endothermic peaks at > 170 °C due to the melting of the HC and SC crystallites. For HC-PLA (L/D = 10/0 and 0/10),  $T_m$  appears at 174 and 177 °C, respectively. In contrast, for SC-PLA, another peak appeared at 227-230 °C, which is attributed to  $T_m$  of the SC crystallites. These results indicate that blending PLLA and PDLA forms both SC and HC domains.

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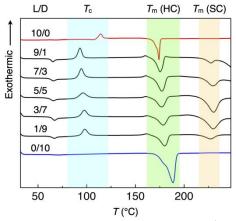


Fig. 3. DSC traces of HC- and SC-PLA with different L/D ratio.

Thermal properties of HC- and SC-PLA are summarized in **Table 1**. From the peak area of  $T_{\rm m}$  in the DSC thermograms, melting enthalpies ( $\Delta H_{m}$ ) of the HC and SC crystallites are determined. By changing the L/D ratio from 10/0 to 5/5,  $\Delta H_{\rm m}$  of HC-PLA ( $\Delta H_{HC}$ ) decreases from 54 to 24 J g<sup>-1</sup> and then increases to 71 J  $g^{-1}$  by further change of the L/D ratio to 0/10. Concomitantly,  $\Delta H_{\rm m}$  of SC-PLA ( $\Delta H_{\rm SC}$ ) significantly increases from 9 to 45 J  $\rm g^{-1}$  and then decreases to 15 J  $\rm g^{-1}$  upon increasing the PDLA content. The  $\Delta H_{HC}$  and  $\Delta H_{SC}$  values show minimum and maximum at L/D of 5/5, indicating that PLA mostly forms SC domains in the 5/5 mixture of PLLA and PDLA.<sup>45</sup> Moreover, the decomposition temperatures ( $T_{\rm dec}$ ) of both HC- and SC-PLA were determined using thermogravimetry differential thermal analysis (TG/DTA, Fig. S2). For pure HC-PLLA,  $T_{\rm dec}$  was at around 280 °C. As the ratio of PDLA to PLLA increased,  $T_{\rm dec}$  increased and reached 337 °C for L/D of 1/9. For pure HC-PDLA,  $T_{\rm dec}$ reached to 354 °C. Therefore,  $T_{\rm dec}$  is mainly affected by  $M_{\rm n}$  of PLA, where large  $M_n$  has high  $T_{dec}$ . However, the TGA profiles of SC-PLA show single decomposition step, indicating that the stereocomplex state is not just a simple mixture of PLLA and PDLA but has a strong interaction between PLLA and PDLA via H-bonding (Fig. 1).

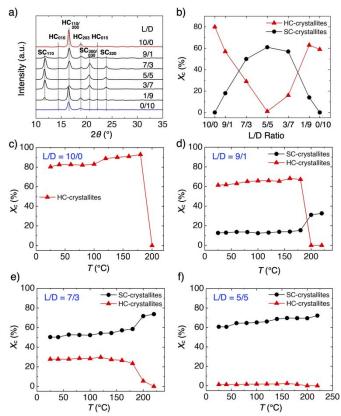
**Table 1.** Summary of the thermal behaviour of HC- and SC-PLA microspheres with different L/D ratio

Sample	$T_{\rm g}$	$T_{\rm c}$	T <sub>m</sub> , HC	$T_{\rm m}$ , SC	$\Delta H_{HC}$	$\Delta H_{SC}$
	(°C)	(°C)	(°C)	(°C)	(J g <sup>-1</sup> )	(J g <sup>-1</sup> )
HC-PLA 10/0	-	109	174	-	54	-
SC-PLA 9/1	65	93	176	227	47	9
SC-PLA 7/3	66	95	175	229	24	42
SC-PLA 5/5	66	96	177	230	24	45
SC-PLA 3/7	67	98	177	229	24	41
SC-PLA 1/9	67	98	179	227	33	15
HC-PLA 0/10	-	-	177	-	71	-

To understand the crystalline states of HC- and SC-PLA, powder X-ray diffraction (PXRD) measurements were conducted. As shown in **Fig. 4a**, powder samples of HC-PLA with L/D of 10/0 and 0/10 show diffraction peaks at  $2\theta = 14.9$ , 16.7, 19.1, and  $22.4^\circ$ , which are assigned as (010), (110)/(200), (203), (015) of the HC crystallite, respectively. In comparison, powder samples with the blend of PDLA and PLLA exhibit three

diffraction peaks additionally at 12.0, 20.8, and 24.1°, which are assigned as (110), (300)/(030), (220) of the SC crystallite, respectively.  $^{25,26,46}$  For the sample with L/D of 5/5, diffractions from the HC crystallite are mostly disappeared. The percentages of the crystallites ( $X_c$ ) of HC- and SC-PLA is calculated from the intensity ratios of the diffraction peaks of the HC and SC crystallites to the entire diffraction. The  $X_c$  value of HC decreases when the L/D ratio changes from 10/0 to 5/5, and then increases when the L/D ratio reaches to 0/10 (**Fig. 4b** blue). Conversely,  $X_c$  of SC increases (decreases) as that of HC decreases (increases) (**Fig. 4b** red).

PXRD measurements were further conducted upon elevating the temperature from 25 to 220 °C (Fig. S3). For the samples with the L/D ratios of 10/0, 1/9, 9/1, and 0/10, the diffraction peaks of the HC crystallite disappear at 200 °C due to the melting of the HC crystallites. For samples with the L/D ratios of 7/3, 5/5, and 3/7, small diffraction peaks from the HC crystallite remain at 200 °C, but they completely disappear at 220 °C. In contrast, diffraction peaks of the SC crystallites remain even at 220 °C. These results are consistent with the DSC results, where T<sub>m</sub> of HC-PLA is around 180 °C while that of SC-PLA is around 230 °C.<sup>47</sup> Figure 4c–f plots the  $X_c$  values versus temperature for the HC and SC crystallites.  $X_c$  of the HC crystallites abruptly drops at 200 °C, and simultaneously, Xc of the SC crystallites increase at 200 °C. These results indicate that the melting of the HC domain subsequently induces the formation of the SC domains.



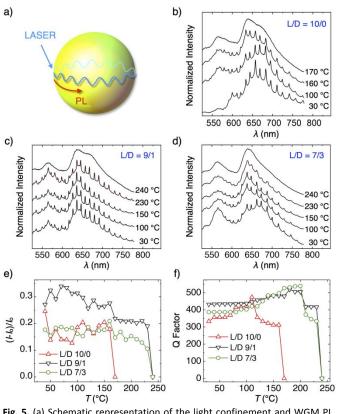
**Fig. 4.** (a) PXRD patterns of HC- and SC-PLA microsphere. (b) The relative degree of crystallinity ( $X_c$ ) of the HC and SC crystallites. (c–f) Relative degree of  $X_c$  versus temperature in the range from 25 to 220 °C for HC and SC crystallites of PLA with the L/D ratio of 10/0 (c), 9/1 (d), 7/3 (e), and 5/5 (f).

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The microspherical structure with smooth surface is advantageous for utilizing as a WGM optical resonator.<sup>48–50</sup> As schematically drawn in Fig. 5a, photoluminescence (PL) generated at the surface of the polymer microsphere is confined via total internal reflection at the medium/air interface and interfere by themselves to show up sharp and periodic resonant PL lines.51-58 To investigate the optical resonator properties of the PLA microspheres, HC- and SC-PLA microspheres are doped with a fluorescent dye, zinc(II) tetraphenylporphyrin (ZnTPP), with  $T_{\rm m}$  ( $^{\sim}$  350 °C) higher than that of PLA. The ZnTPP-doped HC- or SC-PLA microspheres are dispersed on a quartz substrate by a spin coating method. PL spectra of a single HC- and SC-PLA microsphere are measured upon focused laser excitation with 405-nm continuous wave (cw) laser to a single microsphere. 59-63 At 30 °C, PL spectra from a HC-PLA microsphere involves periodic PL lines, attributed to WGMs (Fig. 5b). As shown in Fig. S4, these resonant peaks are assigned as transverse electric (TE) and transverse magnetic (TM) modes. The free spectral range (FSR) of four microspheres with different sizes are plotted in Fig. S5. FSR follows with an equation,

$$FSR = (\lambda^2/n\pi) \cdot (1/d)$$

indicating that the PLA microspheres certainly act as an optical resonator. The clear WGM peaks in the PL spectra is maintained upon thermal heating of the microspheres up to 160 °C (Fig. S6a in more detail). By further heating at 170 °C, these WGM PL peaks disappear due to the melting of HC-PLA that causes the deformation of the microspherical morphology.



**Fig. 5.** (a) Schematic representation of the light confinement and WGM PL. (b–d) PL spectra of a single PLA microsphere after being annealed for 2 s upon photoexcitation at 405 nm with a cw laser. (e) Plots of the normalized PL intensity of the microspheres with L/D = 10/0, 9/1, and 7/3 versus annealing

temperature. (f) Plots of Q factor of the microspheres with L/D = 10/0, 9/1, and 7/3 versus annealing temperature.

The optical resonator properties are observed for microspheres of SC-PLA with L/D of 9/1 and 7/3 (Fig. 5c and d, respectively). Intriguingly, the thermal stability of the microresonator preserves upon thermal heating up to 230 °C. The WGM PL peaks disappear by heating the microspheres at 240 °C (Fig. S6b and c in more detail). In Fig. 5e, temperature dependencies of the WGM peak intensity, normalized by the PL intensity of the background unconfined PL, are plotted. It is obvious that the thermal stability of the optical resonator is higher by ~70 °C for the microspheres of SC-PLA with L/D of 9/1 and 7/3 than that of the HC-PLA microspheres (L/D = 10/0). It is worth noting that the phase transition of PLA affects the shape of PL spectra, possibly caused by the change of the aggregation manner of ZnTPP in the PLA microspheres. We confirm that the ratio of the PL intensity at 603 and 643 nm ( $I_{603}/I_{643}$ ) from a cast film of the PLA microspheres varies at the phase transition temperature (Fig. S7).

To gain insight into the resonant properties in more details, Q factor, defined as the wavelength of the resonant peaks divided by its full width at the half maximum (FWHM), are evaluated.65 Fig. 5f plots Q factor of the PL peaks versus the heating temperature. Q factors gradually increases upon heating, possibly due to the improvement of the surface roughness. However, phase transition causes the loss of Q factors caused by the scattering of the confined light by the crystalline domains. For example, Q factor of the HC-PLA microsphere (L/D = 10/0) dropped from 500 to 360 at 110 °C, where phase transition occurs from glass to the crystalline state. Further drop of Q factor occurs at 170 °C, where melting of the HC-PLA takes place with the collapse of the microspherical morphology. In case of SC-PLA with L/D of 9/1 and 7/3, Q factors drop once at ~200 °C, where HC-PLA crystallites melt and SC-PLA crystallites form, and finally drop off at 240 °C, at which the SC-PLA completely melts.

For comparison, SC-PLA microspheres with L/D of 5/5, 3/7, and 1/9 display quite poor WGM PL (**Fig. S8**), because the surface morphology of the microspheres is rather rough with the RMS roughness greater than 6 (**Fig. 2**). The light confinement is sensitive to the surface roughness, where the rough surface causes a scatter of the confined light, leading to the loss of the optical resonator property.<sup>65</sup> Similarly, HC-PLA microspheres from PDLA (L/D = 0/10) does not show WGM PL due to the ill-defined spherical morphology (**Fig. S8**).

In conclusion, we successfully prepared thermally tolerant optical resonator from stereocomplex crystallites of PLA formed by blending of PLLA and PDLA through oil-in-water miniemulsion method. The SC-PLA microspheres exhibit the higher thermal stability in comparison with the HC-PLA microspheres, where the melting temperature of SC-PLA is more than 50 °C higher than that of HC-PLA. The relative degree of crystallinity of SC-PLA shows maximum when the content of PLLA and PDLA is 5/5, as evaluated by the PXRD studies. The temperature dependent PXRD results clearly show that PLAs maintain the stereocomplex structure even the temperature

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reaches to 220 °C. The temperature-dependent  $\mu$ -PL spectroscopy shows that the SC-PLA resonators exhibit high thermal tolerance in comparison with the HC-PLA resonator, where the WGM resonance properties of SC-PLA preserve even at 230 °C. This work demonstrates the powerful strategy of the stereocomplex formation toward thermally tolerant bio-related materials for optical applications.

### **Conflicts of interest**

There are no conflicts to declare.

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