



Metal Promoted Mo₆S₈ Clusters: A platform for probing ensemble effects on the electrochemical conversion of CO₂ and CO to methanol

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Complete List of Authors:	Perryman, Joseph; University of California Davis Department of Chemistry, Chemistry Ortiz-Rodríguez, Jessica; University of California Davis Department of Chemistry, Chemistry Jude, Joshua; University of California Davis Department of Chemistry, Chemistry Hyler, Forrest; University of California Davis Department of Chemistry, Chemistry Davis, Ryan; Univ. Stanford, SSRL Mehta, Apurva; Univ. Stanford, SSRL Kulkarni, Ambarish; University of California Davis Department of Chemical Engineering and Materials Science Patridge, Christopher; D'Youville College Velazquez, Jesus; University of California, Davis, Chemistry and Chemical Engineering

Conceptual Insights

Selective electrocatalyst materials comprised of Earth-abundant elements have been synthetic targets for years, although a lack of fundamental studies of catalyst electronic structure has hindered the progress of rational material design thus far. In this work, fundamental insights gleaned from electrochemistry and X-ray absorption spectroscopy are utilized in order to elucidate the role of metal promotion on active site geometry and electronic structure. As it relates to electrocatalytic CO₂ reduction, the electronic structures of transition metal sulfide active sites have been tremendously under-investigated experimentally, hence the evidence presented in this work that demonstrates potential tunability of active site functionality may usher a new paradigm in experimental evaluation of energy conversion materials.



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Metal-Promoted Mo₆S₈ Clusters: A platform for probing ensemble effects on the electrochemical conversion of CO₂ and CO to methanol

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Joseph T. Perryman,¹ Jessica C. Ortiz-Rodríguez,² Joshua W. Jude,¹ Forrest P. Hyler,¹ Ryan C. Davis,³ Apurva Mehta,³ Ambarish R. Kulkarni,⁴ Christopher J. Patridge⁵ and Jesús M. Velázquez^{1,6,*}

Presented herein is an investigation of a promising ternary metal sulfide catalyst that is capable of electrochemically converting CO₂ to liquid and gas fuels such as methanol and hydrogen. When promoted by copper, an extended structure of Chevrel-phase Mo₆S₈ clusters is capable of reducing CO₂ and CO to methanol in aqueous conditions with an overpotential of -0.4 V vs RHE. H₂ gas is simultaneously and preferentially evolved during this process, contributing to total current densities as high as 35 mA/cm². It has been observed that Cu₂Mo₆S₈ displays unique catalytic activity in terms of product selectivity, and we attribute this activity to molybdenum sulfide cluster units based on the results of structural, electronic, and electroanalytical characterization. Also discussed is the formulation of an interesting electronic structure-function correlation founded on the basis of X-ray absorption spectroscopic analyses and corroborated by the results of electroanalytical evaluation, where it has been observed that introduction of metal promoting species into the Chevrel-phase framework encourages charge transfer into cluster chalcogen sites.

Introduction

Growing interest in the fields of energy conversion and storage involves developing materials and methods that can facilitate production of liquid fuel products which can easily be incorporated into modern petroleum-based infrastructures.^{1,2} In certain difficult-to-decarbonize energy services such as long-distance shipping, aviation, and production of materials like steel and cement—which combined contribute to over 9.2 Gt of CO₂ emissions per year—implementation of alternative energy sources based on hydrogen or solar energy is infeasible owing to necessarily high gravimetric and volumetric energy densities.³ However, it is immediately apparent that continued production and subsequent combustion of fossil fuels may exacerbate detrimental anthropogenic effects on both public and environmental health. Hence it will be desirable for future methods of fuel production to utilize abundant feedstocks such as solar energy and water, as well as captured and/or atmospheric CO₂.³

This will ensure that future energy cycles are effectively carbon-neutral and based on converted renewable energy.⁴ One potential route towards realizing carbon-neutrality is electrochemical reduction of captured CO₂ to liquid fuels; this process will need to be facilitated by an earth-abundant, selective and efficient catalyst material that is stable under aqueous, near-neutral operating conditions.⁵ Unfortunately, highly selective CO₂ and CO reduction to energy-dense hydrocarbons and oxygenates in water remains a challenge for all known heterogeneous catalysts.⁶

Transition metal chalcogenides are capable of withstanding reductive potentials over wide pH ranges,⁷ and are among the best earth-abundant hydrogen evolution catalysts under extreme pH conditions.^{8,9} Moreover, transition metal chalcogenide materials have been studied for decades, owing to their tunable electronic, structural, and catalytic properties.¹⁰⁻¹⁴ Chevrel-phase (CP) sulfides with formula M_xMo₆S₈ (M = transition metal, alkali metal, x = 0-4) have been the subject of much study due to their high-temperature superconducting behavior as well as their reputation as the first functional multivalent battery cathode materials.¹⁵⁻¹⁷ To the best of our knowledge, despite numerous studies evaluating their performance as hydrodesulfurization, hydrogen evolution, oxygen reduction, and oxygen evolution catalysts, no experimental work has elucidated the ability of Chevrel-phase materials to reduce CO₂.^{8,14,18-20} The chemical composition of catalytically active sites for small-molecule electroreduction is known to have a direct effect on the adsorption strength of reaction intermediates, largely stemming

¹ Department of Chemistry, University of California, Davis, One Shields Avenue, California 95616, United States, Email: jvelazquez@ucdavis.edu.

² Department of Chemistry, University of Puerto Rico, Cayey, Puerto Rico 00736 United States

³ Stanford Synchrotron Radiation Lightsource, SLAC National Accelerator Laboratory, 2575 Sand Hill Road, Menlo Park, California 94025, United States

⁴ Department of Chemical Engineering, University of California, Davis, One Shields Avenue, California 95616, United States

⁵ Department of Chemistry, D'Youville College, Buffalo, NY 14201, United States

⁶ Lead Contact

*Correspondence: jvelazquez@ucdavis.edu

from modulation of electronic structure and subtle alterations in the geometry of binding sites.²¹⁻²⁴ In the context of heterogeneous catalyst surfaces, the former is known as the “ligand effect,” and often, changes to electronic structure through compositional modification lead to accompanying effects associated with unique active-site “ensembles,” wherein neighboring atoms at catalytically active binding sites afford unique coordination interactions for intermediate species with varying energetic favorability.^{25,26} In other words, the presence of multiple elements at an active binding site can afford a distribution of diverse and distinctive coordination environments for reaction intermediates—effectively allowing for independent variation of binding strength for preceding and succeeding adsorbates with relevance in a given catalytic reaction. These phenomena are illustrated graphically in **Figure 1** where intermediate binding over a metallic surface is compared to binding over a surface with a ternary active-site ensemble. Recent computational studies by Liu *et al* suggest that CP sulfides may be promising candidates as selective catalysts for the formation of methanol, both from CO₂ and from syngas (CO and H₂).^{26,27} Specifics of a potential reaction mechanism are discussed later in this work, although it is noteworthy that key intermediate species in the CO₂ reduction reaction (CO₂RR) such as CO* and HCO* are thought to be stabilized at Mo binding sites by cationic promoting species, while adjacent S atoms facilitate sequential hydrogenation toward an energy-dense product.^{26,28} Binding strengths to these two intermediates in particular are often key factors in the rate of CO₂ reduction to liquid fuels; hence it is critical that materials which promote favourable binding are investigated.²⁵

In this work, we report the unique catalytic ability of the Cu₂Mo₆S₈ CP to reduce both CO₂ and CO to methanol without also producing a myriad of other CO₂RR products and provide a rationale for such behavior through interpretation of complimentary electronic and local structure analysis.

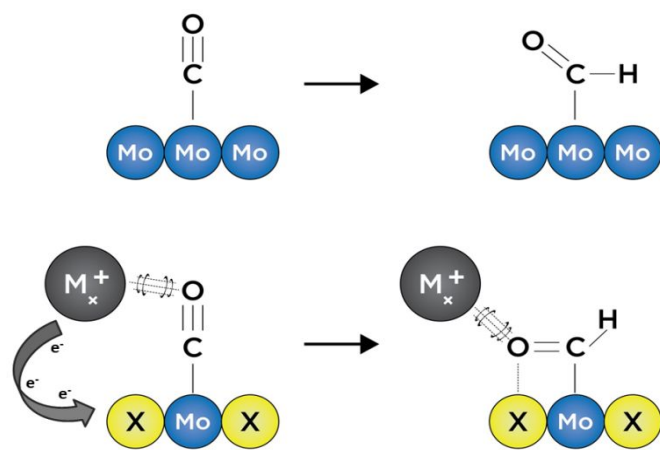


Figure 1. Schematic depiction of CO hydrogenation over a purely metallic surface (top), compared to a promoted metal chalcogenide surface (bottom) where synergistic chalcogen (yellow) inclusion at CO₂RR active Mo sites (blue) encourages a unique intermediate binding geometry. Simplified M_x (grey) Coulombic stabilization of anionic intermediate domains is included for clarity.

Experimental Section

Chemicals and Materials

MoS₂ powder (99%, ~325 mesh), Cu powder (99.995%, ~100 mesh), Ni powder (99.8%, ~325 mesh), Cr powder (99.85%, ~100 mesh), and

Pt mesh (99.99%) were used as purchased from Alfa Aesar. Mo powder (99.99%, ~100 mesh), Na₂CO₃ (99.999% trace metals basis), and NaHCO₃ (>99.5%) were used as purchased from Sigma Aldrich. Na₂CO₃ electrolyte solutions were prepared with 18.2 MΩ/cm water from a Thermo-Fisher Barnstead E-pure® purification system and did not require any pre-electrolysis treatment. Research-grade CO₂ (99.999%) and CO (99.999%) were used as purchased from Matheson Gas. Ag/AgCl reference electrodes were purchased through ALS Japan. Selemion® anion exchange membrane was purchased from AGC Engineering and stored in ultra-pure deionized water prior to use in electrochemical experiments.

Catalyst Synthesis

CP catalysts were prepared by a direct microwave assisted solid-state synthesis method adapted from literature methods.²⁹ To obtain the desired CP, an appropriate metal powder (e.g. Cu, Cr, Ni), Mo powder, and MoS₂ powder were mixed in stoichiometric ratios under N₂ atmosphere, ball-milled for 24 hrs under N₂ and pressed under 25 metric tons over a 20 mm surface area. Compressed powders were transferred to fused quartz tubes, and Al₂O₃ wool was tightly packed above the pellet under +5.0 mbar partial pressure of N₂ as shown in **Figure S1**. This reaction vessel was then transferred to a bath of ~325 mesh graphite powder in a conventional microwave that had been purged with Ar and irradiated with microwaves at a power of 1000 W for 10 minutes. In this process, graphite acts as a microwave susceptor and radiates heat to generate reaction temperatures >900°C after less than one minute of irradiation. After allowing full phase-conversion over the course of a 10 minute reaction time, the quartz tube was removed from the graphite bath and immediately cooled in a room-temperature water bath. In order to obtain bare Mo₆S₈, Cu was chemically etched from a Cu₂Mo₆S₈ pellet according to literature methods which involved bubbling O₂ gas into a 6.0 M HCl solution overnight.³⁰

Structural and Electronic Characterization

Crystal structures and phase purity of as-synthesized CPs were analyzed via powder X-ray diffraction (PXRD) using a Bruker D8 Advance diffractometer with Cu K-alpha radiation (1.5406 Å) as well as at Stanford Synchrotron Radiation Lightsource (SSRL) beamline (BL)2-1 (0.728068 Å). PXRD experiments at BL 2-1 were performed under He atmosphere with a 1.3 Tesla bend magnet and a Si (111) monochromator giving an energy resolution (ΔE/E) of 5x10⁻⁴. Refinement and powder pattern indexing shown in **Figure S2** was performed using the TOPAS analysis software by Bruker.

Catalyst morphology and composition were analyzed before and after electrolysis using an FEI (Hillsboro, OR) 430 Nano Scanning Electron Microscope (SEM) and an FEI Scios DualBeam SEM with an Oxford Energy Dispersive X-ray (EDX) detector, respectively. Further elemental analysis was completed using a PHI Versaprobe 3 X-ray photoelectron spectrometer (XPS) to determine catalyst surface composition and oxidation state before and after electrolysis. Additional structural and electronic information was acquired through ex-situ XAS at SSRL beam lines 4-1 and 4-3 using hard and soft/tender X-rays, respectively, in order to acquire X-ray Absorption Near-Edge Structures (XANES) and Extended X-ray Absorption Fine Structure (EXAFS) information for Cu, Mo, and S.

X-ray Absorption Spectroscopy and Data Analysis

Near-edge (XANES) and fine structure (EXAFS) information was collected SSRL BL 4-1 (molybdenum K-edge and copper K-edge) and BL 4-3 (sulfur K-edge). All samples were collected at room

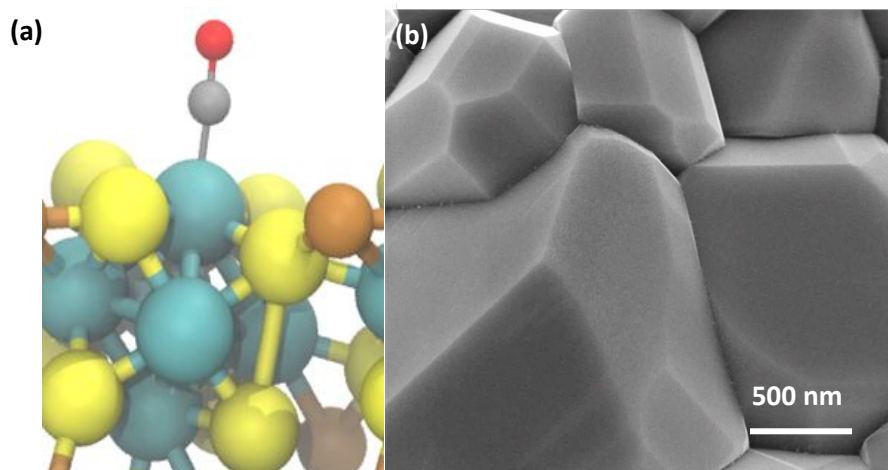


Figure 2. Computationally confirmed interaction between CO and a ternary CP active site ensemble (a). This illustrates that preferential binding of CO on a $\text{Cu}_2\text{Mo}_6\text{S}_8$ CP surface occurs at Mo (blue) active sites, while the proximity of Cu (brown) to S sites (yellow) encourages promoter-to-chalcogen charge transfer. Scanning electron micrograph detailing the faceted morphology of polycrystalline $\text{Cu}_2\text{Mo}_6\text{S}_8$ (b).

temperature (25°C). BL 4-1 operates with a 20 pole 2.0-Tesla wiggler station and uses a liquid N_2 -cooled double-crystal Si (220) monochromator giving an energy resolution ($\Delta E/E$) of 1×10^{-4} and flux of 2×10^{12} . BL 4-3 monochromator uses the Si (111) giving similar resolution and flux. Samples were collected in fluorescence and transmission with foil standards (Mo and Cu) collected simultaneously for calibration, while sodium thiosulfate was measured separately as a calibration standard for sulfur K-edge measurements. Data was calibrated for energy by comparing the first derivative of sample spectrum with that of the respective elemental reference. Scans were performed in triplicate and merged to reduce noise at high k values. Processing was done using Athena and AUTOBK algorithm under the Demeter package.³¹ EXAFS modeling (Mo K and Cu K-edge) was done using Artemis in the Demeter package. Theoretical EXAFS was calculated based on previous refinement for $\text{Cu}_2\text{Mo}_6\text{S}_8$ from literature.^{31–33} Mo K-edge data was fit in R-space using Hanning window and multiple k -weight (1, 2, and 3) using a reciprocal space range of $2.0 \text{ \AA}^{-1} - 12.5 \text{ \AA}^{-1}$ and a real space range of $1.0 \text{ \AA} - 3.2 \text{ \AA}$. The scattering paths were calculated using feff 6.³¹ Dominant scattering paths for the Mo K-edge data, Mo – S (4 paths), Mo – Mo (4 paths), and Mo – Cu (2 paths) were fit using a single energy parameter (E_0), a passive reduction factor (S_0^2) set to 0.83 for Mo K found by fitting the Mo foil standard, bond length shift parameters (drS1, drS2, drMo1, drCu1), and Debye-Waller thermal parameter (ssS, ssMo, and ssCu) giving a total of 8 parameters in the fit. The un-promoted Mo_6S_8 CP was fit with the same scheme without Cu scattering paths. The Cu K-edge data was fit using the data range of $2.0 \text{ \AA}^{-1} - 12.0 \text{ \AA}^{-1}$ in k space and $1.3 \text{ \AA} - 4.2 \text{ \AA}$ in real space. No Cu-Cu scattering paths appeared to contribute significantly to the EXAFS signal and therefore only Cu – S (2 paths), and Cu – Mo (4 paths) were given parameters of E_0 , drS1, drS2, drMo1, ssS, and ssMo. The passive reduction factor (S_0^2) was set to 0.82, found by fitting the Cu foil standard. Theoretical XANES were calculated with Feff9.05 using full multiple scattering.³⁴

Detailed experimental methods related to electrochemical and computational analyses have been included for reference in the supporting information.

Results

Synthesis

The rapid, microwave-assisted solid-state synthetic method implemented in this work has proven capable of yielding highly phase-pure polycrystalline CP materials, including $\text{Ni}_2\text{Mo}_6\text{S}_8$, $\text{Cr}_{1.73}\text{Mo}_6\text{S}_8$, and $\text{Cu}_2\text{Mo}_6\text{S}_8$. In order to observe changes to local and electronic structure with and without a ternary metal promoter atom, the un-promoted Mo_6S_8 CP was also synthesized according to a chemical etching process described in the experimental section. Catalyst morphology can be seen for $\text{Cu}_2\text{Mo}_6\text{S}_8$ in **Figure 2**, while composition can be observed in the EDX scans shown in **Figure S3**. Pawley refinement of synchrotron PXRD information has yielded lattice parameters of $a=b=9.6328 \text{ \AA}$, $c=10.2229 \text{ \AA}$ with a unit cell volume of 821.502 \AA^3 for the $\text{Cu}_2\text{Mo}_6\text{S}_8$ CP of interest in this study. These values are all in close agreement with literature values for the R-3H unit cell.^{16, 35} Results of XRD analysis can be seen in **Figure S4** and **Table S1**.

Local and Electronic Structure

$\text{Cu}_2\text{Mo}_6\text{S}_8$ and Mo_6S_8 CPs were analyzed at the Mo K-edge (20KeV), Cu K-edge (8.9KeV), and S K-edge (2.4KeV) XANES in order to elucidate spectral transitions that yield valuable information regarding frontier orbital population, oxidation state, coordination geometry, as well as charge transfer between species that constitute catalyst active site ensembles. **Figure 3** shows normalized $\mu(E)$ in the K-edge XANES regions for Mo, Cu, and S. **Figure 3a** shows the Cu K-edge spectra for $\text{Cu}_2\text{Mo}_6\text{S}_8$, with a Cu^0 foil for reference. By measuring the position of the most intense peak in the 1st derivative (inflection point) of the spectra compared to the reference foil, we see that the Cu K-edge in $\text{Cu}_2\text{Mo}_6\text{S}_8$ is shifted 2.5 eV higher than the reference. This indicates an oxidized Cu species in $\text{Cu}_2\text{Mo}_6\text{S}_8$ as has been previously determined in literature as well as confirmed in this study by XPS analysis as shown in **Figure S5**. Mo K-edge data is displayed in **Figure 3b** for Mo_6S_8 and $\text{Cu}_2\text{Mo}_6\text{S}_8$ along with a Mo^0 foil for reference. The spectra show near overlap at the edge jump between $\text{Cu}_2\text{Mo}_6\text{S}_8$ and Mo_6S_8 ($\sim \Delta 1 \text{ eV}$), indicating a negligible

change in Mo oxidation state when Cu is introduced to the structure. **Figure 3c** shows the S K-edge data for $\text{Cu}_2\text{Mo}_6\text{S}_8$ and Mo_6S_8 ; a scan for the $\text{Na}_2\text{S}_2\text{O}_3$ reference can be seen in **Figure S6**. There are clear changes to the S electronic structure due to incorporation of the Cu promoter, as will be discussed further later.

After multiple scans of the EXAFS region for all elemental components of $\text{Cu}_2\text{Mo}_6\text{S}_8$ and un-promoted Mo_6S_8 it has been determined that all observed bond lengths are similar to expected values within the uncertainties of the fit. EXAFS information and theoretical fittings plotted in magnitude and real space for Mo and

Cu are shown in **Figure 4(a-d)** where we observe qualitatively different local Mo coordination when promoting species are present compared to un-promoted Mo_6S_8 . Quantitative bond length information extracted from the EXAFS analysis shown in **Figure 4** is discussed in detail below.

Catalyst Evaluation

$\text{Cu}_2\text{Mo}_6\text{S}_8$ was tested in neutral-pH controlled-potential electrolysis experiments with CO_2 dissolved in solution under applied potentials ranging from -1.0 V vs the reversible hydrogen electrode (RHE) to -0.4 V vs RHE, as maximum CO_2RR efficiency occurred within this potential window. We have confirmed that in this potential window, only two liquid-phase CO_2 reduction products are formed after electrolysis; namely, formate and methanol. Nuclear magnetic resonance (NMR) spectra for these CO_2RR products after electrolysis at -1.0 V vs RHE are given in **Figure 5(a,d)**.

When the target molecule was changed from CO_2 to CO in a 0.1 M NaHCO_3 solution, an NMR signal for formate was no longer observed, but the signal for methanol remained. This is evidence that the undesirable formate pathway was entirely suppressed upon removal of CO_2 from the electrochemical cell. This observation is consistent with multiple accounts from literature that report a general lack of formate production when CO is implemented as a reduction feedstock, likely due to the impracticality of switching adsorbate-electrode coordination from C to O, as well as the required insertion of an additional oxygen atom.³⁶ **Figure 5(b,e)** details this result by providing the insets required to visualize formate and methanol production or lack thereof when the electrochemical cell was purged with CO prior to electrolysis. To confirm that observed product formation in this study was the result of CO_2 or CO reduction and not electrolyte interactions, N_2 was purged into a near-neutral pH, 0.1 M NaHCO_3 solution and controlled potential electrolysis was again performed at -1.0 V vs RHE. Notably, during these N_2 control experiments, neither formate nor methanol was observed, as shown in **Figure 5(c,f)**.

As seen in **Figure S7**, it was found that reductive current was dominated by hydrogen evolution. This is evidenced by large geometric current densities for hydrogen evolution compared to those for CO_2RR products. It is worth noting however, that no systematic attempts were made to minimize hydrogen evolution on these CP catalysts through modifications to operating conditions (e.g. ionic liquid electrolyte,³⁷ gas-diffusion methodologies³⁸) or electrochemical cell constructs. In addition to inhibition of the parasitic formate pathway when CO_2 is not present in solution, we see in **Figure S8** a general trend that faradaic efficiencies for methanol conversion from CO increase relative to those from CO_2 at the same potentials. That is, despite lower solubility in water than CO_2 , CO is more efficiently converted to methanol on the $\text{Cu}_2\text{Mo}_6\text{S}_8$ surface although geometric current densities are not entirely dissimilar between CO_2 and CO reduction, indicative that hydrogen evolution remains the predominant reaction over these surfaces.

It was observed that $\text{Cu}_2\text{Mo}_6\text{S}_8$ maintained its electrocatalytic performance over the course of multiple hours of electrolysis, even at the most negative potentials applied. This is evident in **Figure S9** by the stability of the reductive current over time. Further, catalyst stability is evidenced by high-resolution XPS spectra presented in **Figure S5** which indicate no significant changes to the catalyst surface aside from slight oxidation following exposure to moisture and oxygen after the completion of chronoamperometry experiments, illustrated by the evolution of a Mo^{6+} signal in **Figure S10**. **Figure S11** also shows survey scans of the catalyst surface before

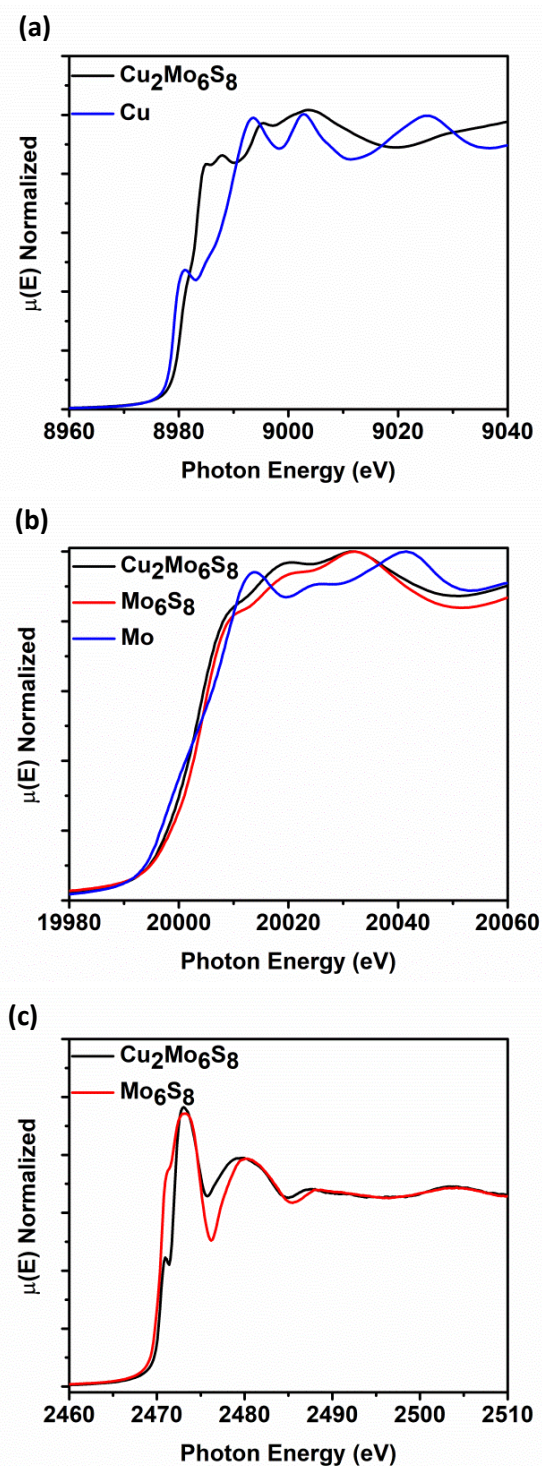


Figure 3. K-edge XANES for Cu in $\text{Cu}_2\text{Mo}_6\text{S}_8$ with a Cu^0 foil for reference (a), K-edge XANES for Mo in $\text{Cu}_2\text{Mo}_6\text{S}_8$ and Mo_6S_8 , with a Mo^0 foil for reference (b), and K-edge XANES for S in $\text{Cu}_2\text{Mo}_6\text{S}_8$ and Mo_6S_8 (c). A reference scan using $\text{Na}_2\text{S}_2\text{O}_3$ is shown in **Figure S6** of the SI. DOI: 10.1039/C5CY01200A, 1-3

and after electrolysis which indicate no plating of foreign metals that could yield false positives of product formation. Lastly, to confirm product formation was not the result of chemical degradation, NMR and GC-TCD analyses were completed on a CO₂-purged Na₂CO₃ electrolyte after Cu₂Mo₆S₈ was left in solution at open circuit for 12 hours. Shown in **Figure S12**, no products were detected after this experiment. Furthermore, H₂S production which would indicate surface degradation during electrolysis was monitored via GC-MS. As shown in **Figure S13**, no H₂S production (m/z = 34) was observed. H₂S was also un-detected qualitatively.

Discussion

Local Coordination and Electronic Structure

CPs have a general structure that includes Mo₆ octahedral clusters surrounded by cubes of S₈, where S atoms lie at each face of the Mo₆ octahedra as depicted in **Figure 6**. These metal-sulfide units extend in a three-dimensional network to form a hexagonal lattice structure. Mo₆S₈ clusters in CPs are tilted in alignment along the ternary axis, with axial S atoms that connect Mo₆S₈ units together in offset chains through relatively short Mo – S bonds. This gives each Mo atom in the bulk network a square pyramidal pseudo-coordination, while Mo atoms exposed on the surface are relatively under-coordinated and exhibit square planar pseudo-coordination.^{19, 39-41} Furthermore, as a result of this arrangement, large cavities exist between cluster units

that are well-suited for occupation by various promoter cations of interest.

The absence of pre-edge features in the Mo K-edge XANES data shown in **Figure 3b**, a signature of s → d-orbital transitions, suggested that the d-orbitals are fully occupied and/or there is little to no orbital mixing between p and d orbitals which indicates octahedral symmetry. The minimal shift in absorption onset and lack of alteration in the features of the near-edge spectra for Cu₂Mo₆S₈ and Mo₆S₈ indicate insignificant change in the electronic structure of the Mo species.⁴² Mo K-edge EXAFS signals and modeling for Mo₆S₈ and Cu₂Mo₆S₈ indicate a dramatic difference in local Mo coordination after Cu incorporation. The Mo₆S₈ has a lone peak (**Figure 4b**) centered around 2.0 Å that originates from Mo – S nearest-neighbor scatterers with an additional small shoulder feature near 2.5 Å. The diminished peak results from a shift in the Mo – S bond distances with axial Mo – S distances increasing to 2.57 Å. Upon Cu incorporation, Mo – Mo bond distances contract towards uniform distances (2.67 Å and 2.73 Å) and constructively interfere to form a new peak. Fitting parameters are included in **Table S2-S4**, while aforementioned bond lengths for dominant/degenerate scatterers are shown in **Table S5**. The fitted bond lengths support small shifts in scattering distances that correlate between the Mo K-edge and Cu K-edge EXAFS modelling in Cu₂Mo₆S₈. These results match closely with those corresponding to a series of magnesiated Mg_xMo₆S₈ compounds that was recently published.⁴³ This previous work by Prendergast *et al* showed that charge compensation during the

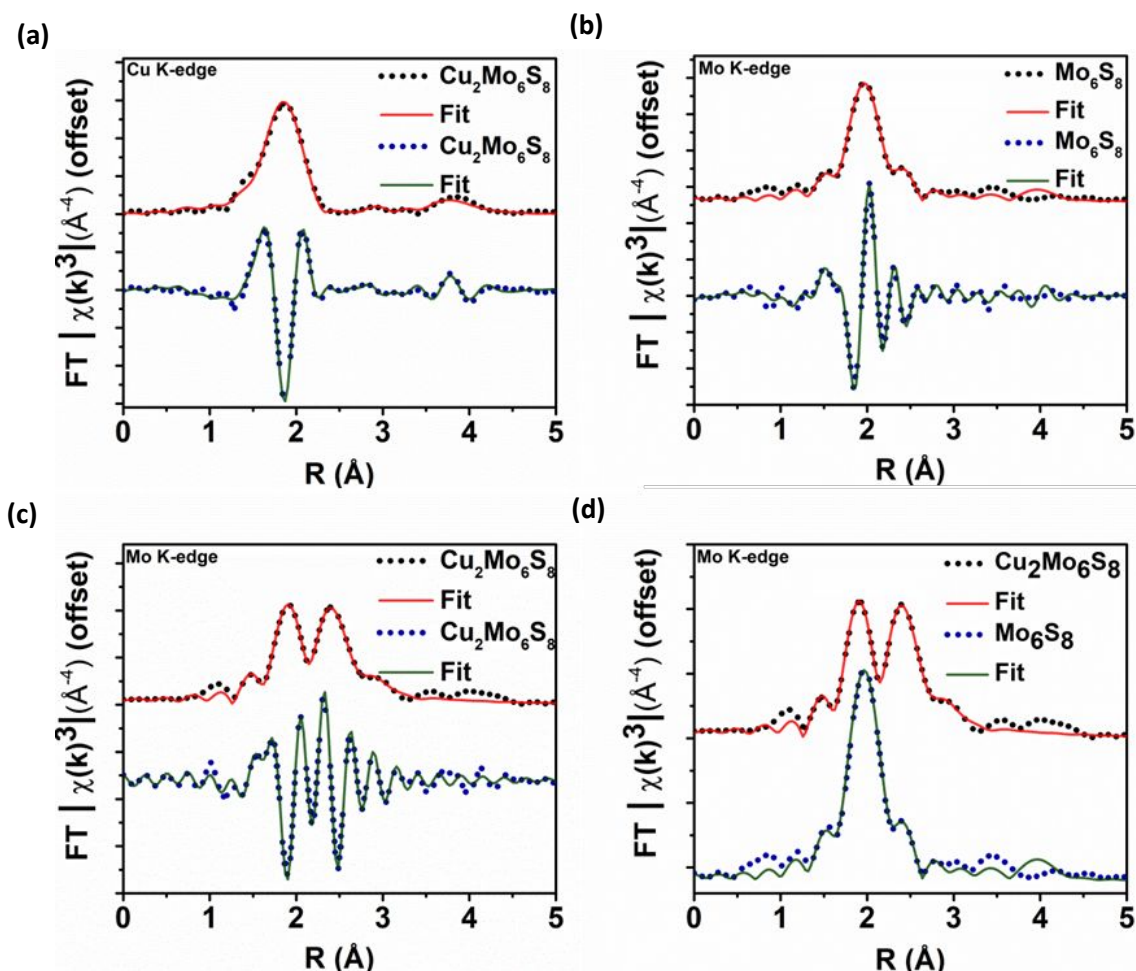


Figure 4. Processed EXAFS information plotted in magnitude and in real space for Cu in Cu₂Mo₆S₈ (a), Mo in Mo₆S₈ (b), Mo in Cu₂Mo₆S₈ (c), and for Mo in Cu₂Mo₆S₈ overlaid with Mo in Mo₆S₈ to compare Mo-S and Mo-Mo signals (d). Qualitative differences in local Mo coordination as presented here are discussed quantitatively in the results and discussion sections.

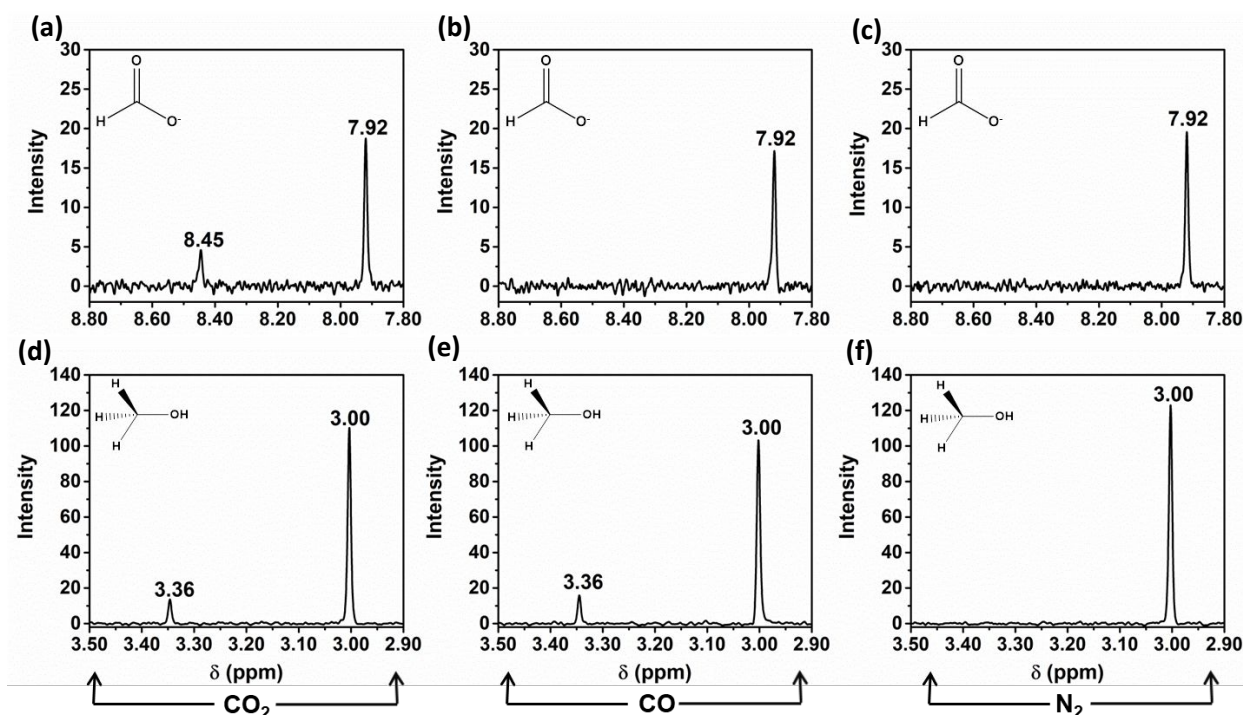


Figure 5. NMR signals for formate at 8.45 ppm (a)–(c) and for methanol at 3.36 ppm (d)–(e) detected after electrolysis under CO_2 (a),(d), CO (b), (e), and N_2 (c),(f). Signals at 3.00 ppm and 7.92 ppm both correspond to the internal standard, *N,N*-dimethylformamide. All displayed NMR spectra are the result of electrolysis at -1.0 V vs RHE in either 0.1M Na_2CO_3 purged with CO_2 or N_2 or 0.1M NaHCO_3 purged with CO .

introduction of an electron donating species actually proceeds through chalcogen atoms—this phenomenon is intriguing because charge compensation during intercalation usually results in monotonic oxidation of transition metal species, although in this case the delocalized electronic nature of the Mo_6 cluster creates low-lying d-orbitals in CPs hence such behavior was not observed for the CP sulfides studied here.^{44–46}

The Mo_6S_8 cluster is deficient by four electrons and is metastable, requiring donation of at least two additional electrons in order for an extended structure to be directly synthesized.⁴⁷ The oxidation state of Cu in $\text{Cu}_2\text{Mo}_6\text{S}_8$ was verified as Cu (I) in earlier studies of the phase, and XPS results (Figure S5) closely match those of Cu_2S .^{32, 48} Hence, incorporation of Cu_2 into Mo_6S_8 induces the donation of two electrons, serving the dual purpose of rendering the framework thermodynamically stable while simultaneously filling available S 3p orbitals and thereby raising the reactivity of the S p-band.²³ This filling of the S p-orbitals can clearly be seen in the S K-edge comparison at 2471 eV between $\text{Cu}_2\text{Mo}_6\text{S}_8$ and Mo_6S_8 in Figure 3c, and is further exemplified graphically in the partial density of states (PDOS) calculations shown in Figure 6b where a clear increase in S 3p PDOS at more negative energies indicates that charge is in fact transferred to S atoms in the structure upon ternary metal incorporation.

X-Ray absorption by S will cause core electrons to excite from 1s \rightarrow 3p orbitals that mix to some degree with Mo 4d orbitals, giving rise to a small pre-edge shoulder that can be clearly seen in the S K-edge XANES spectra at approximately 2471 eV in Figure 3c. As Cu donates electrons to the metal-sulfide cluster, these S orbitals are filled and the pre-edge feature diminishes in magnitude.⁴¹ The feature also appears to sharpen, as fewer transitions are left available for core S electrons. Interestingly, the aforementioned $\text{Mg}_x\text{Mo}_6\text{S}_8$ study shows experimental data that indicates a completely masked pre-edge shoulder for $\text{Mg}_2\text{Mo}_6\text{S}_8$ S K-edge

XANES.⁴³ This is because two Mg atoms would donate a total of four electrons, completing the ideal electron configuration of Mo_6S_8 and eliminating available pre-edge transitions.^{40, 43} In contrast, $\text{Cu}_2\text{Mo}_6\text{S}_8$ only yields two donated electrons (from two Cu^+), hence the shoulder feature is still present and available S 3p orbitals are not completely filled.

To interpret the observed reactivity of CP sulfides in this work, we correlate potentially increased reactivity of the chalcogen species to this filling of 3p orbitals, as shown in the PDOS included in Figure 6b. It is hypothesized that CO hydrogenation will be the rate-limiting step in the CO_2RR to liquid fuels, hence being able to surmount the energy barrier associated with this step will rely strongly on the ability of the catalyst to facilitate hydrogenation. Thus, increasing the population of the S p-band should increase the reactivity of the chalcogen atoms which are thought to play a key role in hydrogenation.^{26, 27}

Effect of Active Site Ensemble on Reactivity

We observed that the $\text{Cu}_2\text{Mo}_6\text{S}_8$ can reduce CO_2 to methanol in aqueous electrolyte at low overpotentials and with only formate production as a competing CO_2RR pathway. Such activity is unique to this particular sulfide catalyst, as no known metal chalcogenides produce methanol in aqueous electrolyte. Interestingly, while copper is one of few transition metals known to reduce CO_2 to methanol (although methane is the preferred C_1 fuel product over copper surfaces),^{49–51} it is not believed that its presence in $\text{Cu}_2\text{Mo}_6\text{S}_8$ is the source of observed catalytic activity. Rather, it is thought that copper merely acts as an electron donor to the catalytically active Mo_6S_8 cluster, rendering the cluster electronically stable and allowing for a direct, high-temperature synthesis of a ternary active-site ensemble similar to the one illustrated in the bottom panel of Figure 1.⁴⁷ In fact, we were

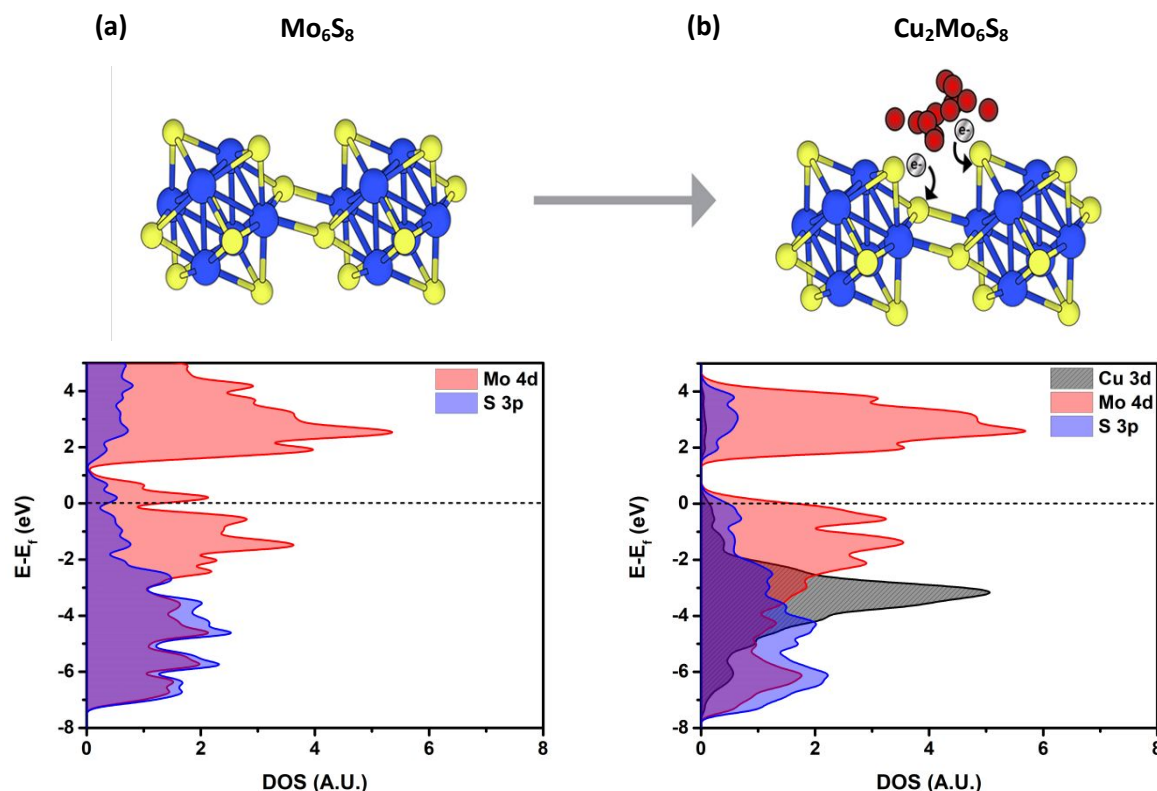


Figure 6. Density of states for Mo_6S_8 (a) and $\text{Cu}_2\text{Mo}_6\text{S}_8$ (b) calculated using the HSE06 functional. A noticeable DOS increase at more negative energies indicates charge transfer to the S atoms as evidenced by XAS. The grey region shows the d10 electronic configuration of Cu(I). Respective structures are represented above where Mo are blue spheres, S are yellow spheres, and Cu are red spheres. Further confirmation of charge transfer to S atoms can be seen in **Figure S18** which shows the result of Bader charge analysis.

able to compute CO binding affinity at Cu sites in the extended structure as being -0.91eV , which is more likely to result in dissociation than prolonged residence on Cu sites for further electroreduction. In contrast, binding energies for CO at Mo sites in $\text{Cu}_2\text{Mo}_6\text{S}_8$ and Mo_6S_8 are much stronger at -1.61eV and -1.50eV , respectively. Additional support for this assertion that copper species do not contribute to observed product formation can be found in **Figure S14**, where we provide NMR spectra that illustrate the production of formate and methanol during controlled potential electrolysis of $\text{Ni}_2\text{Mo}_6\text{S}_8$ and $\text{Cr}_{1.73}\text{Mo}_6\text{S}_8$ (see **Figure S15** for corresponding PXRD patterns for these CPs). Lastly, chronoamperometry was performed for un-promoted Mo_6S_8 to obtain a baseline for CO_2 reduction activity, as shown in **Figure S16**, where methanol is produced at -0.8V vs RHE only when catalyst is present on the carbon paper. It has yet to be determined quantitatively whether the identity and stoichiometry of the metal promoter in a CP lattice has a significant effect on activity or selectivity, although such a phenomenon has been predicted by Liu *et al.*²⁶

Although faradaic yields to CO_2RR products are low as a result of competing hydrogen evolution and mass transport limitations, methanol production efficiency as much as doubled at some potentials when CO was introduced as the target for reduction (**Figure S8**). This is a strong indicator that the pathway for methanol production on CP catalysts does proceed via CO hydrogenation as expected, and further indicates either that the interaction between CO_2 and the Mo_6S_8 units is weaker than between CO and Mo_6S_8 , or that non-polar and sparingly soluble CO_2 does not readily diffuse to

the polarized electrode/electrolyte interface during electrolysis— either (or both) situation may be the case.

The $\text{Cu}_2\text{Mo}_6\text{S}_8$ catalyst investigated here displays among the lowest reported overpotentials for electrochemical methanol production in aqueous media.⁵² Based upon the results of this work, it is hypothesized that CP sulfides promote a reaction pathway that involves CO hydrogenation to methanol, and in order to reach the CO^* intermediate of the CO_2RR , it has been calculated that an associative mechanism occurs which involves the formation of an HOCO^* intermediate as shown in **Figure S17**, followed by an $\text{H}_2\text{O}^*\text{CO}^*$ intermediate. Formation of water through the reverse water-gas shift reaction may improve the energetics associated with a C=O cleavage that is inevitably required for methanol production from CO_2 . This hypothesis seems reasonable, as the Mo d-band center is relatively low-lying with respect to the Fermi level, and is therefore not prone to directly break a C=O bond;^{23, 27} the PDOS results presented here for Mo_6S_8 and $\text{Cu}_2\text{Mo}_6\text{S}_8$ also support this Mo d-band positioning which can also be seen in **Figure 6**. This hypothesis is supported by the production of methanol rather than methane by $\text{Cu}_2\text{Mo}_6\text{S}_8$ in CO_2 and CO saturated solutions.

Kinetic Limitations

It is widely acknowledged that competitive hydrogen evolution often contributes to low overall current efficiencies for the CO_2RR in aqueous electrolytes. Hydrogen evolution in the electrolysis experiments presented here was vigorous enough that nucleation and de-nucleation of H_2 gas may have induced mass-transport difficulties during electrolysis, as the CO_2 target molecule— with

intrinsically low solubility in water (~35 mM for a saturated solution at 1 atm)— may have been unable to diffuse to the electrode-electrolyte interface against such intense gas evolution. To confirm such an assertion, further experimentation utilizing a rotating disk electrode or vapor-fed cell architecture may be required. It is also postulated that a Butler-Volmer trend for product formation was not observed in this study as a result of large currents for competitive hydrogen evolution at more negative applied potentials. This may explain the decrease in Faradaic efficiency for both formate and methanol at more negative potentials.

To further confound the electrochemical reduction of CO₂ to methanol, production of methanol on a CP catalyst likely proceeds through the previously discussed CO hydrogenation pathway in multiple steps. The suspected rate-limiting CO hydrogenation step is likely to involve a large activation energy barrier owing to the high transition state energy required to yield the HCO* intermediate.^{26,27} While extensive intermediate studies have yet to be performed for Cu₂Mo₆S₈, the relatively significant increase in FE towards methanol production when the target CO₂ molecule was replaced by CO suggests that the reaction does proceed via CO hydrogenation, and that the reaction rate is inherently limited by weak adsorption interactions between catalyst active sites and CO₂. Moreover, we believe that the simultaneous production of formate over CP surfaces when CO₂ is the target molecule for reduction is the result of insufficient stabilization of the *HOCO intermediate (predicted by theory) relative to the HCOO* intermediate that follows proton-coupled electron-transfer to *CO₂.^{26,27}

To increase faradaic yields for CO₂RR products compared with the HER, implementation of an electrolyte cation with a greater specific adsorptivity to the electrode surface may serve to tune the electric potential at the outer Helmholtz plane such that proton migration may be suppressed and partially negative atoms within CO₂RR intermediate species may be further stabilized.^{53,54}

Conclusions

Using local probe *ex-situ* XAS, we have observed tuning of the S electronic structure of Mo₆S₈, potentially towards stronger activity for CO hydrogenation. We have also shown that metal promotion in CP frameworks leads directly to modular stabilization of reaction intermediates such as CO, ostensibly through synergistic interactions that include coulombic stabilization by ternary active-site ensembles. This idea has been emphasized in theoretical work by Liu *et al.* for CP systems, and we now have experimental evidence that these hypotheses are valid.

We have demonstrated that Cu₂Mo₆S₈, although previously unexplored as a CO₂ reduction electrocatalyst, is capable of producing methanol—albeit at low partial current densities and low overall efficiency relative to hydrogen evolution. However, this reactivity was observed in aqueous electrolyte, at low overpotentials, and with a unique degree of selectivity relative to many existing electrocatalysts.^{49,52,55} Only two liquid-phase CO₂ reduction products, formate and methanol, were formed during electrolysis on the investigated catalysts. Further, the competing formate pathway was successfully suppressed by the alternative implementation of a CO target molecule, which yielded methanol as the lone liquid-phase product.

It will be of significant fundamental value to incorporate advanced spectroscopic analyses that allow for identification of adsorbed reaction intermediates as a function of applied potential. This avenue of future exploration, in tandem with an *operando* investigation of catalyst surfaces, will allow for reliable determination and rationalization of CO₂ and CO reduction

mechanisms. Promising *operando* techniques such as grazing incidence XAS and XPS will yield necessary mechanistic information by elucidating changes to active-site local coordination and electronic configuration upon application of an applied potential and while in the presence of reduction target molecules dissolved in electrolyte.⁵⁶⁻⁵⁸

Future work is still required to establish the mechanism by which the investigated Chevrel-phase material reduces CO₂ to methanol, as this knowledge will necessarily lead to an increased understanding of the role of active site multifunctionality on reaction efficiency. To develop a catalyst system whose tunable local and electronic structure can affect reactivity and selectivity could be a significant step forward in developing fundamental knowledge in energy conversion and storage research. The unique reactivity displayed by this metal-promoted Mo₆S₈ catalyst family warrants detailed investigation into the precise role of metal promotion in encouraging desirable reaction trajectories over promising energy-conversion catalysts. Hence, future work will elucidate the effect of tunable metal promotion on electronic structure of catalytically active sites such that control over reaction kinetics may be achieved.

Conflicts of interest

There are no conflicts to declare.

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