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# Photoluminescence characteristics and optical temperature sensing of orange-red emitting oxyapatite phosphor $Ca_3La_7(SiO_4)_5(PO_4)O_2:Sm^{3+}$

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In this study, we successfully synthesized a series of oxyapatite phosphors, $Ca_3La_7(SiO_4)_5(PO_4)O_2$ doped with $Sm^{3+}$ ions, using a high-temperature solid-state reaction method. The synthesis process involved varying the concentrations of $Sm^{3+}$ dopants to investigate their impact on the phosphors' structural and luminescent properties. The phase purity of the synthesized material was evaluated using X-ray diffraction (XRD) analysis, which confirmed a hexagonal crystal structure. Additionally, the XRD results validated the high phase purity of the prepared oxyapatite phosphors and demonstrated the successful incorporation of $Sm^{3+}$ ions into two separate crystallographic sites within the lattice. Raman scattering spectra revealed distinct vibrational modes corresponding to the stretching and bending vibrations of the $PO_4$ and $SiO_4$ groups in the structure of the synthesized material. In addition, scanning electron microscopy (SEM) illustrated the morphology of the sample. The photoluminescence excitation and emission spectra of $Ca_3La_7(SiO_4)_5(PO_4)O_2:Sm^{3+}$ phosphors were investigated. Upon excitation with near-ultraviolet (near-UV) light, the phosphors exhibited four distinct emission peaks at wavelengths of 565 nm, 598 nm, 648 nm, and 708 nm. $Ca_3La_7(SiO_4)_5(PO_4)O_2:Sm^{3+}$ phosphors exhibit strong orange-red emission centered at 598 nm, attributed to multipole–multipole interactions, and demonstrate rapid decay rates, making them promising candidates for temperature sensing and supplementary solid-state lighting applications. The CIE chromaticity coordinates of phosphors with varying $Sm^{3+}$ concentrations consistently fall within the orange-red region, confirming their characteristic light emission. Consequently, the synthesized $Ca_3La_7(SiO_4)_5(PO_4)O_2:Sm^{3+}$ phosphor shows significant potential for use in white light-emitting diodes (WLEDs). The relative sensitivity of the $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ was measured to be approximately 0.212% $K^{-1}$ at 293 K, underscoring their potential for temperature sensing applications. This sensitivity is distinct from thermal quenching, which is critical for LED performance. Collectively, these findings demonstrate the suitability of the novel synthesized oxyapatites for applications in solid-state lighting and optical thermometry.

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## 1. Introduction

Temperature has consistently stood out as a widely employed physical measure in both contemporary daily life and scientific investigations, underscoring the crucial need for its precise measurement.[1–4] Precise temperature monitoring plays a crucial role in process control within industrial production and scientific research, particularly in the fields of biomedicine and optoelectronics. High relative thermal sensitivity ($S_r$) has become a crucial indicator in the advancement of luminescent thermometers related to these fields.[5,6] Fluorescence thermometry is a non-contact and non-invasive measurement technique characterized by high spatial resolution, rapid response, and temperature sensitivity.[7] In addition, it is insensitive to fluctuations in the surrounding electromagnetic field and is therefore attracting growing interest in the field of temperature detection.[8] In particular, fluorescence intensity ratio (FIR) technology has been widely adopted for tasks such as microbiological cell detection, high-temperature measurement, and monitoring in manufacturing.[4,9,10] To further advance research in high-temperature sensing, the development of materials with

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exceptional FIR performance is essential. The ability to monitor discernible signals and high temperature sensitivity are all crucial criteria for assessing the excellence of FIR materials.[11,12]

In addition to fluorescence thermometry, white light emitting diodes (WLEDs) have attracted considerable research attention. Today, WLEDs are commercially available, thanks to advancements in high performance, eco-friendliness, long lifespan, and other benefits.[13] However, these WLEDs still struggle to achieve an optimal color rendering index, primarily due to the absence of a red component in their overall composition. To address this issue, europium-based red phosphors have been widely studied due to their superior red light emission properties.[14,15] However, the search for alternative red-emitting ions to replace $Eu^{3+}$ is still increasing due to the need for materials with enhanced performance and broader applicability. Samarium ions ($Sm^{3+}$) have been identified as promising candidates, as they exhibited a narrow red-orange emission under near-UV excitation, making them suitable for solid-state lighting applications.[16–18]

In free $Sm^{3+}$ ions, 4f–4f electric dipole transitions are strictly forbidden due to parity and symmetry selection rules. However, when $Sm^{3+}$ ions are embedded in a crystalline host, the local crystal field disrupts the centrosymmetric environment around the ion, mixing electronic states of opposite parity. This phenomenon, known as forced electric dipole transitions, makes these transitions partially allowed.[19] As a result, the excitation peaks of $Sm^{3+}$-based phosphors align closely with the emission wavelengths of near-UV LED-InGaN chips, enabling effective excitation. This compatibility makes it straightforward to fabricate and integrate $Sm^{3+}$-based phosphors with commercially available LED chips. The $Sm^{3+}$ ion stands out as a favorable option due to its several promising emitting channels in the orange-red spectrum, notably the transitions at $^4G_{5/2} \rightarrow {}^6H_{7/2}$ and $^4G_{5/2} \rightarrow {}^6H_{9/2}$. Consequently, $Sm^{3+}$ has been extensively studied for its potential in the development of phosphors that emit orange-red light.[20–22] In addition, $Sm^{3+}$ ions are well-suited for optical temperature sensing due to their rich set of low-lying excited energy levels. The lower energy states, specifically the $^6H_J$ levels ($J$ = 7/2, 9/2, 11/2, and 13/2), are thermally coupled with the ground state $^6H_{5/2}$.[23] The population distribution among these thermally accessible levels is highly temperature-dependent and follows the Boltzmann distribution,[24] making them ideal for FIR-based thermometry. One of the notable advantages of $Sm^{3+}$ ions is their excellent thermal stability when embedded in an appropriate host lattice. In such hosts, $Sm^{3+}$ can maintain a relatively stable luminescence output even under extended thermal exposure, which is crucial for reliable temperature sensing applications. However, it's important to note that the emission efficiency of $Sm^{3+}$-doped phosphors is still subject to thermal quenching at elevated temperatures. This sensitivity to temperature underscores the importance of choosing a suitable host material that can minimize non-radiative losses and ensure optimal luminescent performance across a wide temperature range.

Among these host materials, oxyapatite phosphors are attracting the interest of researchers due to their considerable advantages in solid-state lighting and optical thermometry, thanks to their high luminescence efficiency, low phonon energy (around 960 $cm^{-1}$), thermal stability with high melting point (typically in the range of 1550–1650 °C), and tunable optical properties.[25] The compound with an oxyapatite structure, represented by the general formula $M_{10}(XO_4)_6O_2$,[26] offers key advantages such as excellent chemical stability and low symmetry, making it a promising candidate for use as the host material in rare earth ion-doped phosphors. The oxyapatite apatite structure is characterized by its compositional flexibility where the metal cation site (M) can accommodate a wide range of elements with different valence states, including monovalent ions such as $Na^+$ and $K^+$, divalent ions such as $Ca^{2+}$, $Sr^{2+}$, and $Ba^{2+}$, and trivalent ions such as rare earth elements (*e.g.*, $La^{3+}$, $Gd^{3+}$, $Y^{3+}$) and $Bi^{3+}$. Similarly, the X site in the crystal framework can be occupied by various high-valence cations such as $P^{3+}$, $Si^{4+}$, or $Ge^{4+}$.[27] This feature allowing it to easily form solid solutions and incorporate a variety of substituent rare earth ions. Previously, many of oxyapatite phosphors such as $Ca_8La_2(PO_4)_6O_2$,[28] $Mn_2Gd_8(SiO_4)_6O_2$,[29] $Sr_2Y_8(SiO_4)_6O_2$,[30] $LiGd_9(SiO_4)_6O_2$,[31] $Sr_2Gd_8(SiO_4)_6O_2$,[32] $Ca_3Gd_7(PO_4)(SiO_4)_5O_2$[33] and others have been widely studied in detail. Detailed studies carried out on these materials have successfully demonstrated their potential as excellent luminescent hosts for diverse optical applications, including solid-state lighting, display technologies and optical thermometry. As a result, ongoing research continues to explore new oxyapatite compositions and dopant combinations, further advancing the development of innovative phosphor materials with optimized luminescent performance.

Inspired by the above considerations, herein, a novel series of oxyapatites-type phosphors, $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$ was synthesized *via* a high-temperature solid-state reaction. The crystal structure, morphology, and luminescence characteristics of the $Sm^{3+}$-doped $Ca_3La_7(SiO_4)_5(PO_4)O_2$ samples were thoroughly examined. Photoluminescence properties under near-UV excitation were evaluated across various $Sm^{3+}$ doping concentrations, revealing intense orange-red emission, which positions these phosphors as promising candidates for WLEDs. Furthermore, temperature-dependent photoluminescence studies were conducted to investigate both the optical properties and the material's potential for temperature sensing applications, encouraging further research in this field.

## 2. Experimental section

### 2.1. Sample synthesis

A series of $Ca_3La_{7-x}(SiO_4)_5(PO_4)O_2$:$xSm^{3+}$ phosphors ($x$ being from 0 at% to 12 at% in the raw materials mixture) were prepared using the solid-state method in an open atmosphere. The starting materials included $CaCO_3$ (99.99%), $La_2O_3$ (99.99%), $SiO_2$ (99.99%), $(NH_4)_2HPO_4$ (99.99%) and $Sm_2O_3$ (99.99%) from Sigma Aldrich, were systematically weighed according to the stoichiometric molar ratio. The materials were then transferred to an agate mortar and ground for several hours. The mixture was then placed in an aluminum crucible and heated to 1173 K for 240 minutes in a muffle furnace. The powder obtained was cooled, finely ground and calcined at

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1423 K for 480 minutes in a muffle furnace. Finally, the powder was cooled to room temperature.

The production of these phosphors was based on the following chemical reaction:

$$3CaCO_3 + \frac{7-x}{2}La_2O_3 + 5SiO_2 + (NH_4)_2HPO_4 + \frac{x}{2}Sm_2O_3 \rightarrow Ca_3La_{7-x}Sm_x(SiO_4)_5(PO_4)O_2 + \text{by-products.}$$

### 2.2. Characterization

Structural analysis of the synthesized $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$xSm^{3+}$ powders was realized by X-ray diffraction (XRD). Measurements were performed using a Bruker-AXS D8-Advance diffractometer equipped with a vertical q–q goniometer and Cu Kα radiation (1.5406 Å). A position-sensitive detector Lynx-Eye-XE-T with an aperture angle of 2.94° was employed to detect the diffracted X-rays. In the Rietveld refinement process, data were recorded with a step size of 0.02° and a step duration of 2 seconds in the $2\theta$ range from 5° to 80°. Silicon (Si(510)) was chosen as the sampling medium due to its low background. Rietveld refinement calculations were performed using Match3! software. Raman spectra were acquired using a Renishaw InVia confocal Raman microscope, equipped with an edge filter, a 50× objective and an argon ion laser ($Ar^+$) used for excitation (at a wavelength of 514 nm). A JEOL 1011 scanning electron microscopy (SEM) was used to examine the morphology of the powders. To determine their composition, electron dispersive X-ray spectroscopy (EDS) was used, performed with an environmental scanning electron microscope (ESEM, FEI Quanta 600) coupled to an Oxford Inca 3.0 microanalyzer.

Photoluminescence (PL) spectra under 404 nm excitation were measured. This 404 nm-blue excitation source was obtained using fundamental femtosecond pulses provided by a Kerr-lens mode-locked (KLM) Ti:sapphire laser which produces tunable pulses across the spectral range of 700–900 nm with up to 2 W of average power and a repetition rate of 76 MHz. The generated fundamental pulses have a pulse duration 0 f–150 fs. The blue laser pulses are achieved using second harmonic generation of the fundamental pulses in a $BiB_3O_6$ nonlinear crystal that is specially cut and AR-coated for this purpose. A spectral analyzer (OSA, Ando AQ6315-E) was used for detection. Temperature-dependent photoluminescence assessments were conducted in a temperature range spanning from 293 K to 513 K, facilitated by a custom-made heating device in which very high-resolution thermocouple sensor was used for temperature monitoring. Photoluminescence excitation (PLE) spectra were measured by means of a spectrofluorometer (FS5, Edinburgh Instruments) fitted with an internal xenon lamp for excitation. Measurements of luminescence lifetime were carried out employing a photomultiplier tube (R928P, Hamamatsu) installed in a temperature cooling box (CoolOne, Horiba). The luminescence wavelength was chosen through the use of a dual monochromator (Bentham, DTMS300).

## 3. Results and discussion

### 3.1. Phase purity, crystal structure and morphology

XRD measurements were used to confirm the phase purity of the prepared samples. XRD patterns of the host $Ca_3La_7(SiO_4)_5(PO_4)O_2$ and $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$ at% $Sm^{3+}$ ($2 \leq x \leq 12$) are shown in Fig. 1(a). These oxyapatites show similar diffractograms and correspond well to pure apatite-type structures crystallizing in the hexagonal system with space group $P6_3/m$. It can be noted that all diffraction peaks can be basically indexed to the corresponding standard data for the hexagonal phase of oxyapatite $Ca_2La_8(SiO_4)_6O_2$ (CCDC file number 1922998, also available in the ICSD database under file number 140674) (see Fig. 1(b)), with no significant signals from impurities observed, even at $Sm^{3+}$ concentrations of up to 12 at%. The sample therefore remains single-phase, and the doped ions have not caused any substantial changes in the crystal structure. Crystalline formations of the $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ were examined using Rietveld refinements and the results are shown in Fig. 2 and Table 1. Additionally, Table 2 provides a summary of the atomic coordinates, site locations, occupancy factors (O.F), and isotropic temperature factors ($B_{iso}$) determined for $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ sample.

Based on the data obtained to determine the crystal structure of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$, a perspective view along the $a$–$b$ crystallographic plane with a double-cell projection along the $b$ axis was generated, as shown in Fig. 3(a). Two distinct categories of cationic sites have been distinguished, called Ca/La(I) and Ca/La(II). The 4f Ca/La(I) site is nine-coordinated with $C_3$ symmetry, and the 6h Ca/La(II) site is seven-coordinated with $C_s$ symmetry.[34–36] The cations were linked by isolated $PO_4/SiO_4$ tetrahedra. The oxide ion O4 did not belong to any $PO_4/SiO_4$ tetrahedra, is known as free oxygen[37] and had the ability to form a short, high covalent bond with the 6h site cation.[38] $Sm^{3+}$ dopant ions are likely to replace trivalent $La^{3+}$ ions in the host matrix, in two types of sites with different point symmetries, $C_3$ (with an oxygen coordination number of IX) and $C_s$ (with an oxygen coordination number of VII), as shown in Fig. 3(b).

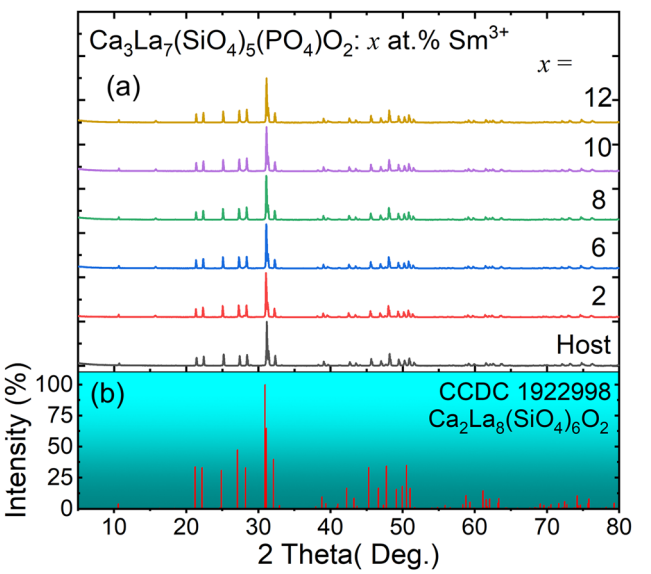


Fig. 1 (a) XRD patterns of $Ca_3La_7(SiO_4)_5(PO_4)O_2$ with different doping concentrations of $Sm^{3+}$ ions (0–12 at%); (b) standard powder diffraction card of $Ca_2La_8(SiO_4)_6O_2$; reference plot was downloaded from CCDC database, file number 1922998 (ICSD 140674).


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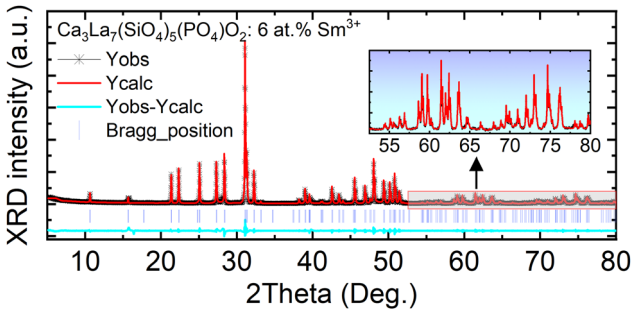


Fig. 2 Rietveld refinement plots for $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$: the experimental profile-black line; the calculated pattern-red line; the difference plot-blue line and the Bragg position-blue vertical line; (inset zoomed view in the range between 52 and 80°).

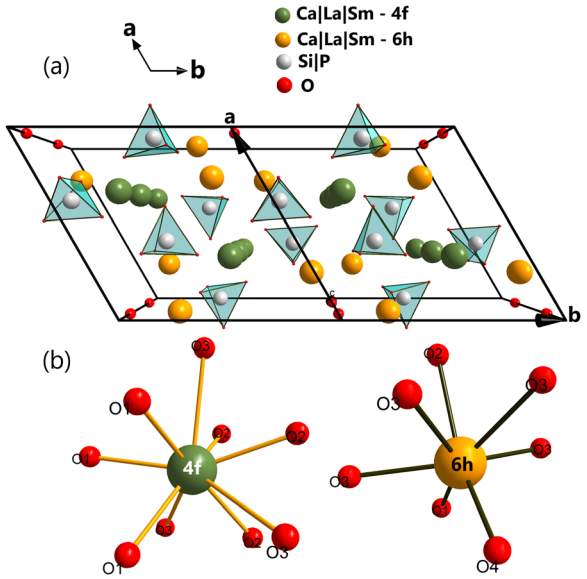


Fig. 3 Crystal structure of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$: (a) perspective view along the *a*–*b* crystallographic plan with double cells projection along *b* axis; (b) oxygen coordination environment of the two cationic crystallographic sites 4f and 6h.

The ionic radii of the $Sm^{3+}$ ions are $r_{Sm}(IX) = 1.123$ Å and $r_{Sm}(VII) = 1.02$ Å, which are smaller than those of the $La^{3+}$ cations making up the host, $r_{La}(IX) = 1.216$ Å and $r_{La}(VII) = 1.1$ Å.[39] This demonstrates the reduction in cell volume caused by the incorporation of $Sm^{3+}$ ions into the host matrix. The volume of the undoped host is 569.311 $Å^3$,[40] whereas the volume of the doped $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ is reduced to 568.778 $Å^3$.

The average crystallite size of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$ phosphors was calculated using Debye–Scherrer equation:[41]

$$D = \frac{K\lambda}{\beta \cdot \cos\theta}, \quad (1)$$

where $D$ is the average crystallite size, $K = 0.9$ is the Scherrer constant, $\lambda = 1.54188$ Å is the wavelength of CuKα radiation, $\beta$ is the full width at half maximum (FWHM) of the diffraction peak (in radians), and $\theta$ is the corresponding Bragg angle (in radian). Based on the (211) reflection ($2\theta \approx 31.1°$ for the undoped sample), the calculated crystallite size for various $Sm^{3+}$ doping concentrations of 0, 2, 6, 8, 10, and 12 at%, yielding values of 65.7, 64.3, 65.5, 62.7, 63.4 and 63.9. These results indicate that the crystallite size remains relatively stable and does not exhibit a significant variation with increasing $Sm^{3+}$ content.

Table 1 Refinement parameters of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$

| Parameters | Value |
|---|---|
| Sample | $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ |
| Crystallographic class | Hexagonal |
| Space group | $P6_3/m$ |
| Laue group | $6/m$ |
| Cell formula units ($Z$) | 1 |
| Space group number (setting number) | 176 (1) |
| Lattice parameters | $a = b = 9.614(1)$ Å $c = 7.1047(1)$ Å |
| Lattice volume | 568.778 $Å^3$ |
| Diffraction radiation type (wavelength) | Cu Kα (1.54188 Å) |
| Measurement temperature | 295 K |
| Software | Match 3 (crystal impact-demo version) |
| Reliability factors | $X^2 = 5.87$ $R_{wp} = 18.10$ $R_p = 10.80$ |

Table 2 Atomic coordinates and structure parameters of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$; (O.F = occupancy factors, $B_{iso}$ = isotropic temperature factors)

| Atoms | Wyck. | $x$ | $y$ | $z$ | O.F. | $B_{iso}$ |
|---|---|---|---|---|---|---|
| Ca1\|La1\|Sm1 | 4f | 1/3 | 2/3 | −0.0032(1) | 0.448\|0.519\|0.033 | 2.002(1) |
| Ca2\|La2\|Sm2 | 6h | 0.2318(5) | −0.0131(2) | 1/4 | 0.036\|0.908\|0.056 | 1.963(3) |
| Si\|P | 6h | 0.4068(3) | 0.3773(1) | 1/4 | 0.893\|0.107 | 1.234(2) |
| O1 | 6h | 0.3221(5) | 0.4929(4) | 1/4 | 1 | 2.423(5) |
| O2 | 6h | 0.5987(1) | 0.4705(8) | 1/4 | 1 | 2.964(1) |
| O3 | 2a | 0 | 0 | 1/4 | 1 | 1.234(2) |
| O4 | 12i | 0.3465(4) | 0.2623(1) | 0.0728(1) | 1 | 1.254(1) |

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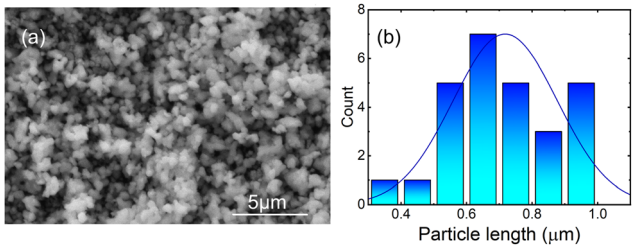


Fig. 4 (a) SEM image of the $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ phosphor; (b) the corresponding particle size distribution.

The morphology and size distribution of the synthesized $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ phosphor were analyzed using scanning electron microscopy (SEM), as shown in Fig. 4(a). The image reveals spherical shaped, agglomerated particles with dimensions in the micrometer range. As illustrated by the size distribution in Fig. 4(b), the particles exhibit a narrow size distribution, centered around ~0.65 μm. Although some variation in particle size is observed, the overall distribution remains consistent and the particles are morphologically uniform. Elemental analysis of the synthesized sample $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ was carried out by EDS analysis (Fig. 5(a)). This analysis confirmed the presence of Ca, La, Si, P, O, and Sm elements, supporting the elemental composition of the $Ca_3La_7(SiO_4)_5(PO_4)O_2$:Sm phosphor. However, phase formation was confirmed through XRD analysis, as discussed earlier. Notably, the concentration of Sm is relatively low, indicating that the prepared $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6% $Sm^{3+}$ sample is a single-phase material. The analysis of the element distribution map depicted in Fig. 5(b) suggests an even distribution of all elements across the surface of the studied area.

### 3.2. Raman spectroscopy

Fig. 6 shows the spectra obtained at room temperature, covering a range from 200 to 1200 $cm^{-1}$. These spectra have been

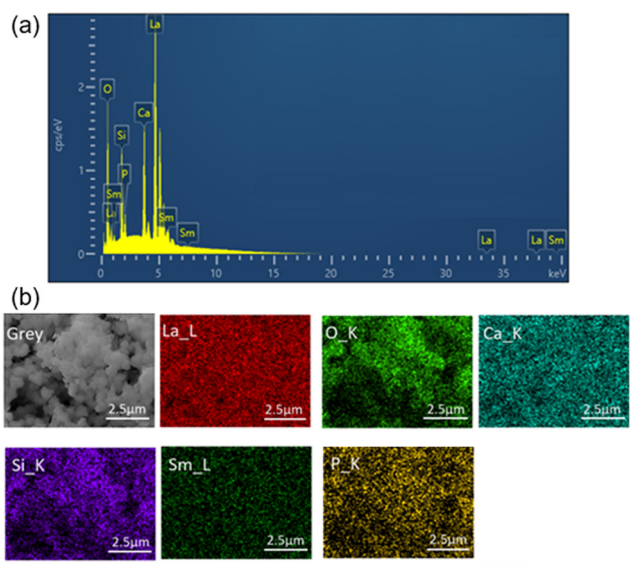


Fig. 5 (a) EDS image and (b) elements mapping analysis of the $Ca_3La_7(SiO_4)$-$(PO_4)O_2$:6 at% $Sm^{3+}$ phosphor.

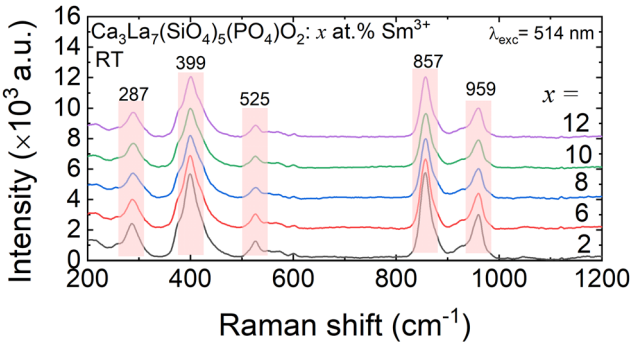


Fig. 6 Raman scattering spectra for $Ca_3La_7(PO_4)_5(SiO_4)O_2$:$x$ at% $Sm^{3+}$ ($x$ = 2, 6, 8, 10 and 12).

compared with recognized references for apatitic phases from previous research[42] in order to discern the vibrational bands associated with ($PO_4^{3-}$) and ($SiO_4^{4-}$) groups. The vibrational modes associated with ($PO_4^{3-}$) groups were observed only at 959 $cm^{-1}$, representing symmetrical vibrational modes ($\upsilon_1$). The ($SiO_4^{4-}$) groups exhibit characteristic peaks at 857 $cm^{-1}$ and 399 $cm^{-1}$, attributed to symmetrical ($\upsilon_1$) and symmetrical ($\upsilon_2$) vibrational modes, respectively. In addition, there are weaker lines at 525 and 287 $cm^{-1}$, which can be attributed to the antisymmetric ($\upsilon_4$) and symmetrical ($\upsilon_2$) deformation modes, respectively. The maximum phonon energy measured for $Ca_3La_7(SiO_4)_5(PO_4)O_2$:Sm was absorbed at 959 $cm^{-1}$.

### 3.3. Luminescence properties of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$$Sm^{3+}$ phosphors

The PLE and PL emission spectra of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$$Sm^{3+}$ (where $x$ = 2, 6, 8, 10, and 12 at%) phosphors are shown in Fig. 7(a) and (b), respectively. The excitation spectrum displays distinct sharp peaks, which are attributed to the partially allowed f–f forbidden transitions of $Sm^{3+}$.[43,44] Specifically, the excitation peaks observed at 344, 361, 375, 389, 404, 424, 447, 461, 475, and 489 nm correspond to the following electronic transitions: $^6H_{5/2} \rightarrow {}^4K_{17/2}$, $^6H_{5/2} \rightarrow {}^4D_{3/2}$, $^6H_{5/2} \rightarrow {}^4D_{1/2}$, $^6H_{5/2} \rightarrow {}^4L_{15/2}$, $^6H_{5/2} \rightarrow {}^6F_{7/2}$, $^6H_{5/2} \rightarrow {}^4I_{15/2}$, $^6H_{5/2} \rightarrow {}^4M_{19/2}$, $^6H_{5/2} \rightarrow {}^4F_{5/2}$, $^6H_{5/2} \rightarrow {}^6I_{13/2}$, and $^6H_{5/2} \rightarrow {}^6I_{11/2}$ of $Sm^{3+}$, respectively. It is clear that the most prominent peak occurs at 404 nm, corresponding to the $^6H_{5/2} \rightarrow {}^4F_{7/2}$ transition, signifying strong absorption in the near-UV region. This same 404 nm peak is also utilized to monitor the emission spectra in the range of 525 to 750 nm. Additionally, the excitation spectrum closely matches that of GaN-based LED chips in the near-UV range (350–420 nm), suggesting its potential as an effective material for GaN-based LED applications in the near-UV spectrum.[43,45]

The PL spectra of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$$Sm^{3+}$ ($x$ = 2, 6, 8, 10 and 12 at%), synthesized by the solid-state reaction method, are shown in Fig. 7(b) for an excitation wavelength of 404 nm. These spectra exhibit four distinct and intense emission peaks, appearing at around 565 nm, 598 nm, 648 nm and 708 nm, corresponding to the following electronic transitions of $Sm^{3+}$ ions: $^4G_{5/2} \rightarrow {}^6H_{5/2}$, $^4G_{5/2} \rightarrow {}^6H_{7/2}$, $^4G_{5/2} \rightarrow {}^6H_{9/2}$, and $^4G_{5/2} \rightarrow {}^6H_{11/2}$,


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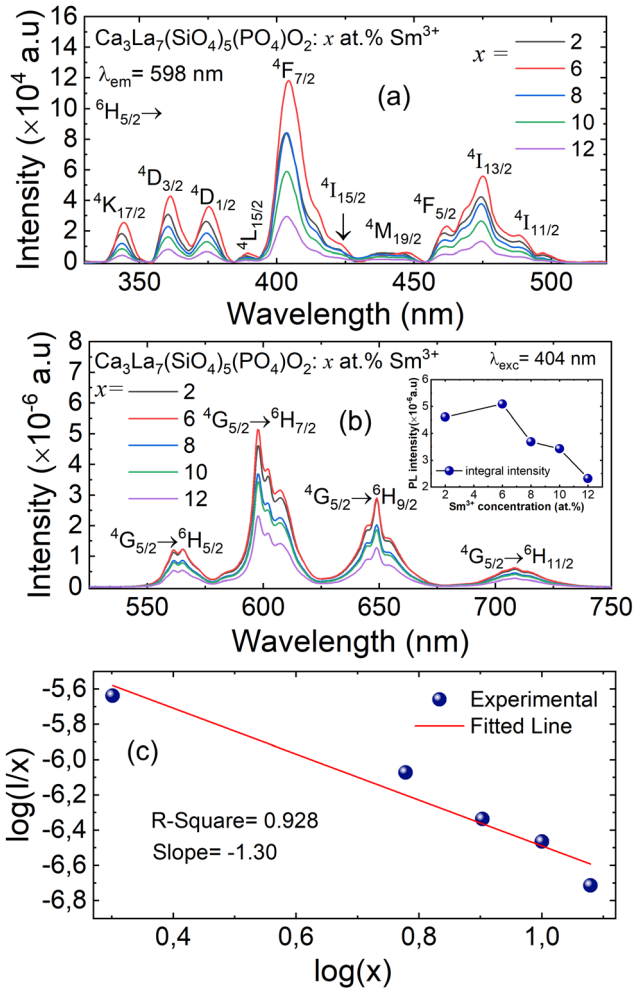


Fig. 7 (a) The PL excitation spectra (PLE; $\lambda_{em}$ = 598 nm); (b) photoluminescence (PL; $\lambda_{exc}$ = 404 nm) spectra of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$Sm$^{3+}$ ($2 \le x \le 12$ at%) phosphors, and inset shows the relationship between PL intensity and Sm$^{3+}$ concentration; (c) fitting line of log($x$) on log($I/x$) in $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$Sm$^{3+}$ phosphors beyond the quenching concentration.

respectively. These transitions are characteristic of Sm$^{3+}$ and align with previously reported data.[46]

The most prominent emission band, located at 598 nm, is attributed to the $^4G_{5/2} \rightarrow {}^6H_{7/2}$ transition of Sm$^{3+}$ ions. This transition is characteristic of red-orange emission, resulting from a combination of forced magnetic dipole (MD) and electric dipole (ED) transitions.[45] Moreover, the $^4G_{5/2} \rightarrow {}^6H_{9/2}$ transition occurring at 648 nm is associated with an electric dipole (ED) transition. Additionally, the $^4G_{5/2} \rightarrow {}^6H_{5/2}$ transition results in a band at 565 nm, which is attributed to a magnetic dipole (MD) transition.[47] We detect a weak emission around 708 nm, attributed to the $^4G_{5/2} \rightarrow {}^6H_{11/2}$ transition, which is involved in defining the color index of LED.[48]

Under the 404 nm excitation wavelength, the emission intensity of Sm$^{3+}$ ions in $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$Sm$^{3+}$ phosphors is profoundly influenced by Sm$^{3+}$ ion concentration (see the inset Fig. 7(b)). Emission intensity increases progressively with Sm$^{3+}$ concentration, reaching a maximum at $x$ = 6 at%. Consequently, the optimum concentration of Sm$^{3+}$ ions in the $Ca_3La_7(SiO_4)_5(PO_4)O_2$ matrix was found to be 6 at%. Above this concentration ($x$ > 6 at%), emission intensity begins to decrease due to a phenomenon known as "concentration quenching". The phenomenon of concentration quenching occurs because, at higher concentrations of doped ions Sm$^{3+}$, the distance between these ions decreases. This proximity leads to increased non-radiative energy loss, causing a gradual reduction in luminous intensity.[49] The critical distance ($R_c$) between the doped ions plays a crucial role in characterizing the concentration quenching mechanism. If $R_c$ is equal to or less than 5 Å, the concentration quenching mechanism is attributed to exchange interactions. Conversely, when $R_c$ exceeds 5 Å, it is associated with electrical multipolar interactions. The evaluation of $R_c$ can be accomplished by employing a specific equation to identify the fundamental interaction mechanism.[50]

$$R_c = 2\left(\frac{3V}{4\pi Z x_c}\right)^{\frac{1}{3}}, \qquad (2)$$

In this context, $V$ represents the volume of the unit cell, $Z$ represents the number of cationic sites available for Sm$^{3+}$ ions in the unit cell, and $x_c$ indicates the atomic fraction of the activator at which quenching occurs. For the $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% Sm$^{3+}$, the values are $Z$ = 1, $V$ = 568.778 Å$^3$, and $x_c$ is 0.06 for the emission peak at 598 nm. Applying the given formula, the $R_c$ for Sm$^{3+}$ ions in $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% Sm$^{3+}$ was calculated to be 26 Å, which exceeds 5 Å. This result suggests that exchange interactions are not the dominant quenching mechanism in the Sm$^{3+}$-doped $Ca_3La_7(SiO_4)_5(PO_4)O_2$.

To gain deeper insight into the energy transfer (ET) mechanism responsible for the luminescence quenching observed in $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$Sm$^{3+}$ phosphors at higher doping concentrations, Dexter and Schulman's theoretical framework is employed:[50]

$$\frac{I}{x} = k\left[1 + \beta(x)^{\frac{\theta}{3}}\right]^{-1}, \qquad (3)$$

Here, $k$ and $\beta$ represent fixed values specific to the host lattice concerned. $I$ represents the emission intensity and $x$ the Sm$^{3+}$ concentration. The parameter $\theta$ serves as an indicator of the multipolar electrical nature. In this context, $\theta$ values of 6, 8, and 10 correspond to electric dipole–dipole (d–d), electric dipole–quadrupole (d–p), and electric quadrupole–quadrupole (q–q) interactions, respectively. When the concentration $x$ surpasses the critical quenching level ($x_c$), eqn (2) can be roughly converted into eqn (4) when the value of $\beta(x)^{\theta/3}$ significantly exceeds 1:

$$\log\left(\frac{I}{x}\right) = A - \frac{\theta}{3}\log x, \quad \text{where } A = \log k - \log \beta, \qquad (4)$$

As shown in Fig. 7(c), the relationship between the logarithm of ($I/x$) and the logarithm of $x$ exhibits an almost linear behavior. The slope of the fitted linear trend was found to be −1.30, and the corresponding interaction parameter $\theta$ was calculated to be 3.9, a value that is reasonably close to 6. This indicates that nonradiative ET between Sm$^{3+}$ ions taking place *via* d–d type interaction for the concentration quenching of

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$Sm^{3+}$ ions in $Ca_3La_7(SiO_4)_5(PO_4)O_2$ phosphors. This mechanism becomes increasingly significant at higher $Sm^{3+}$ concentrations, where the average distance between activator ions is reduced, thereby facilitating more efficient energy migration through dipole–dipole coupling.

To better understand the luminescence mechanism of $Sm^{3+}$ ions in the $Ca_3La_7(SiO_4)_5(PO_4)O_2$ host matrix, Fig. 8 illustrates a simplified energy level diagram of $Sm^{3+}$, highlighting the key excitation and emission transitions, as well as the possible non-radiative cross-relaxation (CR) pathways involved. As shown in the excitation section of the diagram, when $Sm^{3+}$ ions absorb photons (excited at 404 nm), electrons are promoted from the ground state ($^6H_{5/2}$) to various higher-lying excited states. However, these excited electrons do not always directly return to the ground state through radiative emission. Instead, they typically undergo a series of non-radiative relaxations (NR), eventually populating the lower $^4G_{5/2}$ excited state. Once electrons accumulate in the $^4G_{5/2}$ state, they relax radiatively to several lower-energy states of the $Sm^{3+}$ ion, particularly to $^6H_{5/2}$, $^6H_{7/2}$, $6H_{9/2}$, and $^6H_{11/2}$. These radiative transitions are responsible for the characteristic orange-red emissions observed in the photoluminescence spectra. In addition to the radiative pathways, the diagram also identifies CR channels, which represent non-radiative energy transfer processes between neighboring $Sm^{3+}$ ions.[51] These mechanisms become more prominent at higher doping concentrations and can lead to quenching effects by depleting the population of emitting states. Specifically, cross-relaxation can occur when an excited $Sm^{3+}$ ion transfers part of its energy to a neighboring ion in the ground state, resulting in both ions settling into intermediate energy levels without photon emission.[51] The possible transitions contributing to cross-relaxation in $Sm^{3+}$-doped $Ca_3La_7(SiO_4)_5(PO_4)O_2$ are: $^4G_{5/2} \rightarrow {}^6F_{5/2} = {}^6H_{5/2} \rightarrow {}^6F_{11/2}$ (CR1), $^4G_{5/2} \rightarrow {}^6F_{7/2} = {}^6H_{5/2} \rightarrow {}^6F_{9/2}$ (CR2), $^4G_{5/2} \rightarrow {}^6F_{9/2} = {}^6H_{5/2} \rightarrow {}^6F_{7/2}$ (CR3) and $^4G_{5/2} \rightarrow {}^6F_{11/2} = {}^6H_{5/2} \rightarrow {}^6F_{5/2}$ (CR4).

The decay curves for $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$$Sm^{3+}$ doped with varying concentrations of $Sm^{3+}$ ions ($x$ = 2, 6, 8, 10, and 12 at%) are shown in Fig. 9(a). The data clearly demonstrate that decay times progressively decrease with increasing $Sm^{3+}$ concentration. For samples with low $Sm^{3+}$ concentrations, the decay curves are well approximated by a single exponential function. However, at higher $Sm^{3+}$ concentrations, the decay curves deviate from a single exponential behavior. This deviation is attributed to non-radiative energy transfer processes between $Sm^{3+}$ ions.[52] Under these conditions, the average lifetime of $Sm^{3+}$ can be calculated using the following equation:

$$\tau_m = \frac{\int_0^\infty tI_{(t)}\mathrm{d}t}{\int_0^\infty I_{(t)}\mathrm{d}t}, \quad (5)$$

where $I(t)$ represents the luminescent intensity at time ($t$), the initial intensity being normalized to 1. The average lifetimes ($\tau_m$) of the $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$$Sm^{3+}$ phosphors were measured for various $Sm^{3+}$ doping concentrations ($x$ = 2, 6, 8, 10, and 12 at%), yielding values of 1.69, 1.03, 0.87, 0.66, and 0.59 ms, respectively. As illustrated in Fig. 9(b), the luminescence decay curves become progressively shorter and exhibit a non-exponential behavior with increasing $Sm^{3+}$ concentration. This trend can be attributed primarily to CR processes occurring between neighboring $Sm^{3+}$ ions. As the doping level increases, the average distance between $Sm^{3+}$ ions decreases, enhancing the likelihood of non-radiative energy transfer between closely spaced ions. In such CR processes, an excited $Sm^{3+}$ ion in the $^4G_{5/2}$ state can transfer part of its energy to a nearby $Sm^{3+}$ ion in the ground state $^6H_{5/2}$, resulting in both ions transitioning to intermediate energy states, thus bypassing radiative emission. This energy migration mechanism leads to a faster depopulation of the excited state and, consequently, a reduction in the observed lifetime.

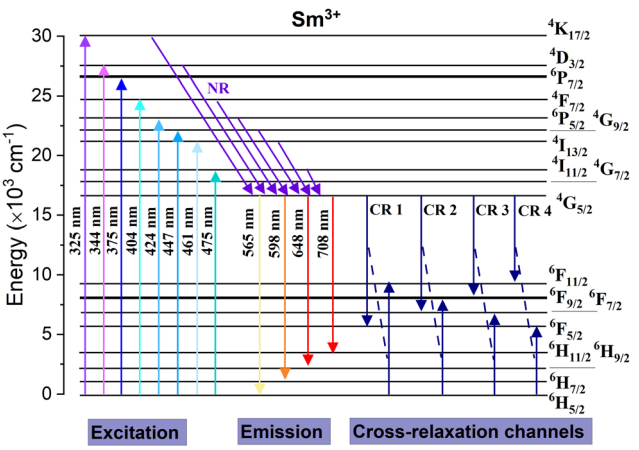


Fig. 8 Energy level scheme showing excitation and emission transitions in $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$ phosphors and the possible cross-relaxation channels; NR – non-radiative relaxation; CR – cross-relaxation channel.

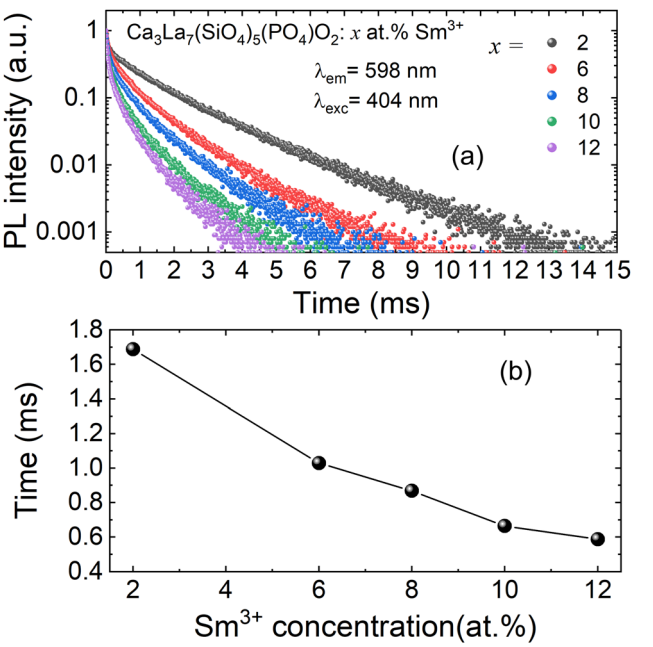


Fig. 9 (a) Decay curves of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$$Sm^{3+}$ ($x$ = 2, 6, 8, 10, and 12 at%) ($\lambda_{exc}$ = 404 nm, $\lambda_{em}$ = 598 nm); (b) plot of the lifetime of $Sm^{3+}$-doped matrix phosphor $Ca_3La_7(SiO_4)_5(PO_4)O_2$ as a function of $Sm^{3+}$ concentration.


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### 3.4. Thermal stability and optical temperature sensing

The temperature-dependent emission response of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ phosphors under 404 nm excitation was systematically investigated over the temperature range of 293–513 K (Fig. 10(a)). This study aimed to evaluate the thermal stability of $Sm^{3+}$ emission and to assess the material's potential for temperature-sensing applications. The emission of the $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ phosphor remained unchanged in terms of wavelength positions, while the intensity showed a continuous decrease owing to the thermal quenching effect.[53] Solid-state lighting systems, such as LEDs, typically operate at around 430 K. At this temperature, the emission intensity of the $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ phosphor retains 94.2% of its initial intensity at RT. In comparison with reported $Sm^{3+}$-doped phosphors, the emission intensities recorded around this temperature for $Ca_3GdAl_3B_4O_{15}$,[54] $CaLa_4Ti_4O_{15}$,[55] $CaGd_2(MoO_4)_4$,[56] and $KLa_{0.97}Sr_3(PO_4)_3F$[57] are 90.8%, 81.6%, 70.9%, and 85.5%, respectively. These results demonstrate that the $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$ phosphor exhibits superior thermal stability compared to many other $Sm^{3+}$-doped hosts. Even at 493 K, the emission intensity shows only a modest decrease, retaining 86.1% of its initial value. To obtain a more comprehensive analysis of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$, the activation energy ($E_a$) was determined. The intensity of the 598 nm ($^4G_{5/2} \rightarrow {}^6H_{7/2}$) emission peak was analyzed using the Arrhenius relationship:[58]

$$I(T) \approx \frac{I_0}{1 + c\exp\left(-\frac{E_a}{kT}\right)}, \quad (6)$$

where the initial intensity represented by $I_0$, the intensity at a specific temperature $T$ denoted by $I(T)$, a constant $c$, the activation energy for thermal quenching denoted by $E_a$, and the Boltzmann constant denoted by $k$, eqn (6) can be easily rearranged as such:

$$\ln\left[\left(\frac{I_0}{I}\right) - 1\right] = -\frac{E_a}{kT} + \ln c. \quad (7)$$

By analyzing the gradient of eqn (7), we can determine the activation energy ($E_a$). To obtain the value of $E_a$, a graph is created by correlating $\ln[(I_0/I) - 1]$ with $1/kT$, giving a linear relationship with a slope of $-E_a$. As shown in Fig. 10(b), this plot shows a linear pattern, with a slope of around −0.226. As a result, for the phosphors synthesized in this work, we calculate that $E_a$ has a value in agreement with previous results in phosphors doped with $Sm^{3+}$ cations such as $Ba_2SiO_4$:$Sm^{3+}$ (0.2707 eV),[53] $YVO_4$:$Sm^{3+}$ (0.17 eV),[59] NaKLa(Nb, Ta)$O_5$:$Sm^{3+}$ (0.22 eV),[60] $Mg_3Y_2Ge_3O_{12}$:$Sm^{3+}$ (0.20 eV).[46]

In the temperature range studied (293 K to 513 K), we observed clear differences in the luminescence intensity attributed to temperature changes for the three $Sm^{3+}$ emission peaks. These emission peaks correspond to the $P1$ transition ($^4G_{5/2} \rightarrow {}^6H_{5/2}$) and the $P2$ and $P3$ transitions, which are Stark sublevels of the $^4G_{5/2} \rightarrow {}^6H_{7/2}$ transition, as demonstrated in Fig. 10(a). For a more comprehensive analysis of the properties of the synthesized phosphors as a function of temperature, Fig. 10(c) and (d) show the intensities of the $P1$, $P2$ and $P3$ emission peaks. $P1$ exhibits relatively stable fluorescence intensity with temperature variations, while $P2$ and $P3$ display only slight changes. This observation highlights the importance of analyzing the fluorescence intensity ratios (FIRs) for $P1/P2$ and $P3/P2$ to better understand the thermometric performance of the material.

Furthermore, literature research has shown that the thermometric luminescence performance of the synthesized phosphors, in this study $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$ at% $Sm^{3+}$, requires the

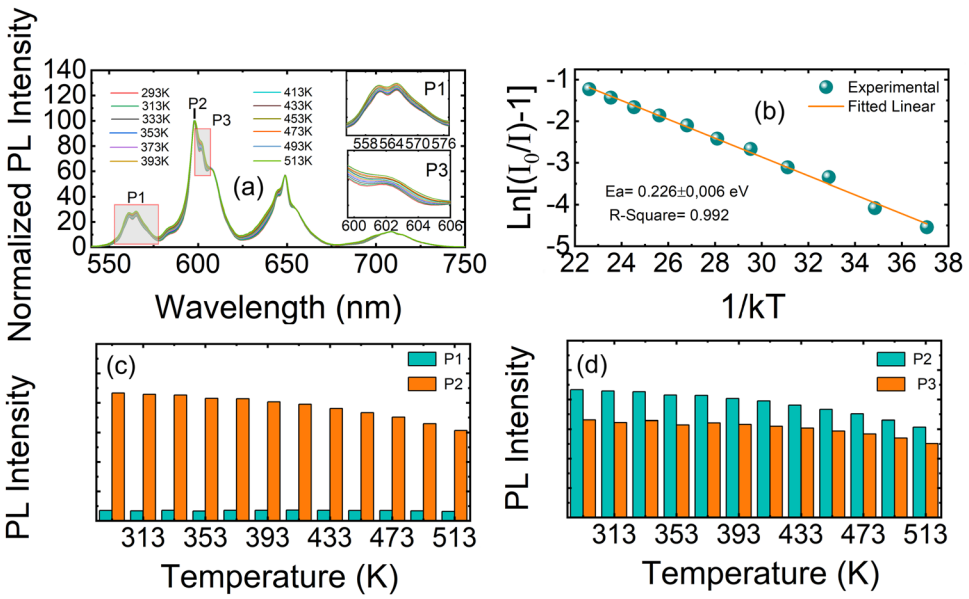


Fig. 10 (a) Emission spectra as a function of temperature (293–513 K) of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ under 404 nm excitation; (b) plot of $Ln[(I_0/I) - 1]$ *versus* activation energy $1/kT$ for thermal quenching of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ for the 598 nm ($^4G_5/_2 \rightarrow {}^6H_7/_2$) transition; the bar graph shows the evolution of (c) $P1$ and $P2$, (d) $P2$ and $P3$ for $Ca_3La_7(SiO_4)_5(PO_4)O_2$ doped with 6% $Sm^{3+}$ as a function of temperature.

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determination of four specific parameters. These parameters are as follows:[61]

(I) Fluorescence intensity ratio, represented by FIR = $P1/P2$ and FIR = $P3/P2$, which serve as thermometric parameters.

(II) Absolute thermal sensitivity ($S_a$) in Kelvin ($K^{-1}$).

(III) Relative thermal sensitivity ($S_r$) in Kelvin for a variation of 100% ($K^{-1}$ %).

(IV) Temperature uncertainty ($\delta T$).

Firstly, the thermometric parameters FIR($P1/P2$) and FIR($P3/P2$) were determined by calculating the ratios of the fluorescence intensities of the emission peaks centered at the corresponding wavelengths. In a broader context, temperature dependence can be modeled following the approach reported by Brites *et al.* for lanthanide thermometers.[62] This model is based on the fact that the total transition probability of an emission level is the sum of the non-radiative and radiative transition probabilities,[63] and by relating the luminescence intensity to the inverse of the total transition probability,[64] the temperature dependence of the intensity ratio will be expressed as follows:

$$\mathrm{FIR}(P1/P2) = \frac{P1}{P2} = \Delta_{01}\frac{1+\sum_i \alpha_{1i}\exp\left(-\frac{\Delta E_{1i}}{kT}\right)}{1+\sum_i \alpha_{2i}\exp\left(-\frac{\Delta E_{2i}}{kT}\right)}, \quad (8)$$

$$\mathrm{FIR}(P3/P2) = \frac{P3}{P2} = \Delta_{02}\frac{1+\sum_i \alpha_{3i}\exp\left(-\frac{\Delta E_{3i}}{kT}\right)}{1+\sum_i \alpha_{2i}\exp\left(-\frac{\Delta E_{2i}}{kT}\right)}, \quad (9)$$

where $T$ is the absolute temperature; $k$ is the Boltzmann constant; 1, 2 and 3 represent the three emissions whose intensities are used to estimate thermometric performance; $\Delta_{01}$ is the ratio $P01/P02$ at 0 K for emissions 1 and 2; $\Delta_{02}$ is the ratio $P03/P02$ at 0 K for emissions 2 and 3. $\alpha_{1i}$, $\alpha_{2i}$ and $\alpha_{3i}$ correspond to the ratio between the non-radiative and radiative probabilities for emission levels 1, 2 and 3, respectively; and the sign of the sum varies from $i = 1$ to $n$, where $n$ are all the possible deactivation channels for the non-radiative processes of intensity transitions $P1$, $P2$ and $P3$. Finally, $\Delta E_{1i}$, $\Delta E_{2i}$ and $\Delta E_{3i}$, expressed in $cm^{-1}$, are the energy differences between the thermally coupled Stark sublevels of transitions 1, 2, and 3. These parameters reflect the influence of thermal redistribution among Stark sublevels on the observed FIRs. Fig. 11(a) illustrates the FIR values represented by the ratio ($P1/P2$), while Fig. 11(b) depicts the FIR values denoted by ($P3/P2$). These FIRs demonstrate how thermal redistribution among Stark sublevels contributes to the observed intensity ratios.

In situations where the exponential term dominates the transition intensities, as observed in this work, and assuming a single dominant non-radiative deactivation channel ($1 \ll \alpha_i \exp(-\Delta E_i/kT)$), eqn (8) and (9) simplify to reflect these relationships:

$$\mathrm{FIR}(P1/P2) = B1\exp\left(\frac{\Delta E_2 - \Delta E_1}{kT}\right) = B1\exp\left(\frac{-C1}{T}\right), \quad (10)$$

$$\mathrm{FIR}(P3/P2) = B2\exp\left(\frac{\Delta E_2 - \Delta E_3}{kT}\right) = B2\exp\left(\frac{-C2}{T}\right), \quad (11)$$

where $\Delta E_1$, $\Delta E_2$, and $\Delta E_3$ are the Stark sublevel energy differences for transitions $P1$, $P2$, and $P3$, and $B1$ and $B2$ are constants obtained from

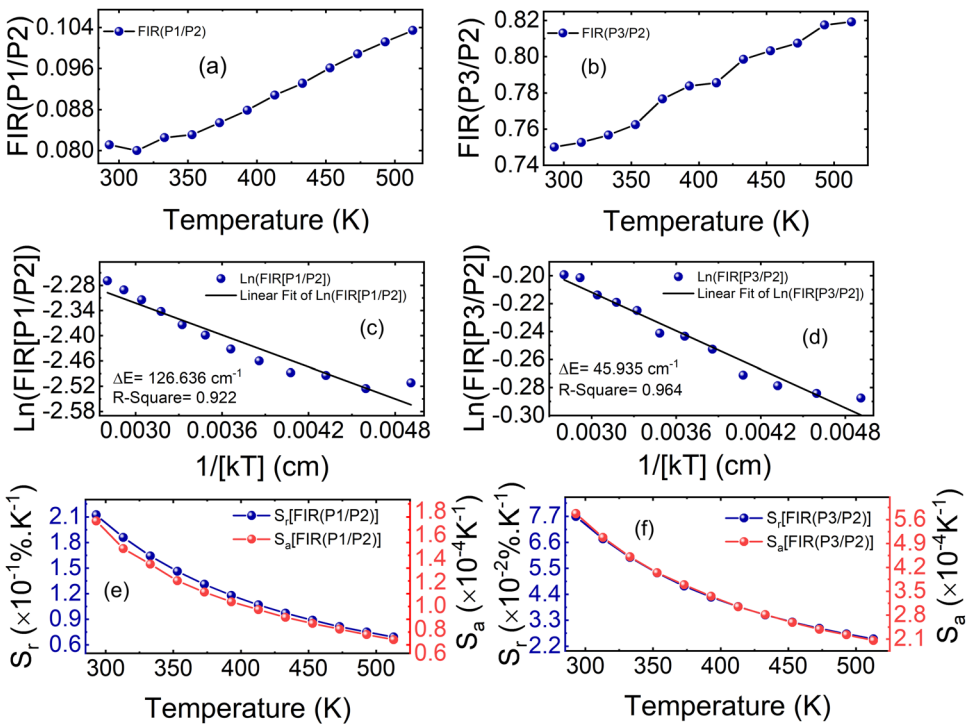


Fig. 11 (a) FIR($P1/P2$) values and fitting curve; (b) FIR values ($P3/P2$) and fitting curve, (c) Ln(FIR[$P1/P2$]) as a function of 1/[$kT$]; (d) Ln(FIR[$P3/P2$]) as a function of 1/[$kT$]; (e) $S_a$ and $S_r$ curves of (FIR[$P1/P2$]); and (f) $S_a$ and $S_r$ curves of (FIR[$P3/P2$]).


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the fitting process. The coefficients $C1 = \frac{\Delta E_2 - \Delta E_1}{k}$ and $C2 = \frac{\Delta E_2 - \Delta E_3}{k}$ quantify the energy disparity driving the thermal redistribution among these sublevels. In this context, we have used a fitting procedure of eqn (9) and (10) to match the experimental relationship between emission and temperature in the sample examined $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$. We have estimated the values of *B*1, *B*2, *C*1 and *C*2. They are respectively equal to 0.143, 0.929, 182.240 and 66.073 respectively.

Next, we plot Fig. 11(c) and (d), showing Ln(FIR[*P*1/*P*2]) and Ln(FIR[*P*3/*P*2]), respectively, as functions of 1/[*kT*] to determine the activation energy differences ($\Delta E_2 - \Delta E_1$ and $\Delta E_2 - \Delta E_3$) using the linear fitting method. These values, derived from the slopes of the fitted lines, correspond to 126.636 $cm^{-1}$ and 45.935 $cm^{-1}$, respectively. This result highlights distinct thermal quenching mechanisms influencing the luminescence behavior, which are essential for achieving reliable and accurate temperature sensing applications. The larger activation energy difference from the FIR(*P*3/*P*2) curve enables high sensitivity within specific temperature ranges, while the smaller activation energy difference broadens the overall detection scope. These complementary activation energy differences allow the fluorescence intensity ratio method to achieve both precise and versatile luminescent thermometry. Consequently, $Sm^{3+}$-doped $Ca_3La_7(SiO_4)_5(PO_4)O_2$ phosphors prove highly effective for applications requiring both high precision and wide temperature coverage, underscoring their potential in advanced thermal sensing technologies.[65]

Furthermore, to evaluate the sensitivity of subsequent detection and determine the thermometric efficiency of our materials, it is crucial to calculate two key parameters, $S_a$ and $S_r$. Based on eqn (10) and (11) for the phosphors $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$, we can determine the absolute thermal sensitivity $S_a$, which was first established by dos Santos *et al.* in 1998 and is expressed as follows:[66]

$$S_a[\mathrm{FIR}(P1/P2)] = \left|\frac{d[\mathrm{FIR}(P1/P2)]}{dT}\right|, \quad (12)$$

$$S_a[\mathrm{FIR}(P3/P2)] = \left|\frac{d[\mathrm{FIR}(P3/P2)]}{dT}\right|. \quad (13)$$

In general, the relative thermal sensitivity $S_r$ is regarded as a reference for evaluating and comparing the efficiency of various thermometers, whatever their type. On the other hand, $S_a$ is not suitable for comparing the performance of different luminescent thermometers, as it depends on factors such as experimental setup, lifetimes, and absorption characteristics.[62] $S_r$ means the greatest change in the FIR intensity ratio for each degree of temperature. $S_r$ values for FIR(*P*1/*P*2) and FIR(*P*3/*P*2) can be determined using the equations provided:[67]

$$S_r[\mathrm{FIR}(P1/P2)] = \left|\frac{1}{\mathrm{FIR}(P1/P2)} \times \frac{d[\mathrm{FIR}(P1/P2)]}{dT}\right| \times 100\%, \quad (14)$$

$$S_r[\mathrm{FIR}(P3/P2)] = \left|\frac{1}{\mathrm{FIR}(P3/P2)} \times \frac{d[\mathrm{FIR}(P3/P2)]}{dT}\right| \times 100\%, \quad (15)$$

Following eqn (12)–(15), $S_a$ and $S_r$ of $Ca_3La_7(SiO_4)5(PO_4)O_2$: 6 at% $Sm^{3+}$ in the temperature range 293–513 K are calculated and presented in Fig. 11(e) and (f) for FIR(*P*1/*P*2) and FIR(*P*3/*P*2), respectively. Both parameters decrease noticeably with increasing temperature, highlighting the dependence of sensitivity on the thermal environment.

For FIR(*P*1/*P*2), the optimum values for $S_a$ and $S_r$ are $1.722 \times 10^{-4}$ $K^{-1}$ and 0.212% $K^{-1}$ respectively at 293 K. On the other hand, for FIR(*P*3/*P*2), the optimum values for $S_a$ and $S_r$ are $5.775 \times 10^{-4}$ $K^{-1}$ and 0.077% $K^{-1}$ respectively at 293 K. In addition, evaluation of the temperature uncertainty ($\delta T$) is a crucial factor in assessing the material's temperature detection, and its value can be calculated using the following equations:[68]

$$\delta T[\mathrm{FIR}(P1/P2)] = \frac{1}{S_r[\mathrm{FIR}(P1/P2)]} \times \frac{\delta[\mathrm{FIR}(P1/P2)]}{\mathrm{FIR}(P1/P2)}, \quad (16)$$

$$\delta T[\mathrm{FIR}(P3/P2)] = \frac{1}{S_r[\mathrm{FIR}(P3/P2)]} \times \frac{\delta[\mathrm{FIR}(P3/P2)]}{\mathrm{FIR}(P3/P2)}, \quad (17)$$

where $\delta$(FIR/FIR) represents the relative error parameter, and its value generally oscillates around 0.5% in FIR measurements.[69] The temperature uncertainty ($\delta T$) for FIR(*P*1/*P*2) and FIR(*P*3/*P*2) is shown in Fig. 12(a) and (c), respectively. The substantial relative sensitivity enabled us to estimate a maximum temperature resolution of around 1.751 K ($T$ = 550 K) for the $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6% $Sm^{3+}$ phosphors. These results suggest that the synthesized phosphors $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$ represent a promising thermosensitive material with commendable temperature resolution.

$Sm^{3+}$-doped phosphors are recognized for their susceptibility to thermal degradation. To assess the temperature-dependent repeatability of FIR, a crucial aspect linked to the prolonged utilization of phosphors for temperature sensing, a cyclic heating test of the $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6% $Sm^{3+}$ phosphors was conducted between 293 and 513 K, as depicted in Fig. 12(b) and (d). The results from the two cycles indicate promising repeatability of 98%. Compared with previously reported $Sm^{3+}$-doped phosphors, the optimum $S_r$ value obtained for $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$ is relatively modest: for instance, YAG:$Sm^{3+}$ has shown sensitivity as high as 0.5033% $K^{-1}$ at 500 K,[70] while $Ca_2LaNbO_6$:$Sm^{3+}$ exhibits a comparable value of 0.22% $K^{-1}$ at 313 K.[71] In contrast, other hosts such as LSYW have achieved significantly higher sensitivities, reaching up to 1.58% $K^{-1}$ at 301 K.[72] Such variations in relative sensitivity are influenced significantly by the host matrix composition, the energy levels involved, and the selection of thermally coupled levels. The performance of the present material remains adequate for practical temperature sensing applications, particularly in systems where extremely high sensitivity is not required. The relatively lower sensitivity can be compensated by the advantages offered by the host material, including good thermal stability, reproducibility, and a simple excitation/emission scheme.

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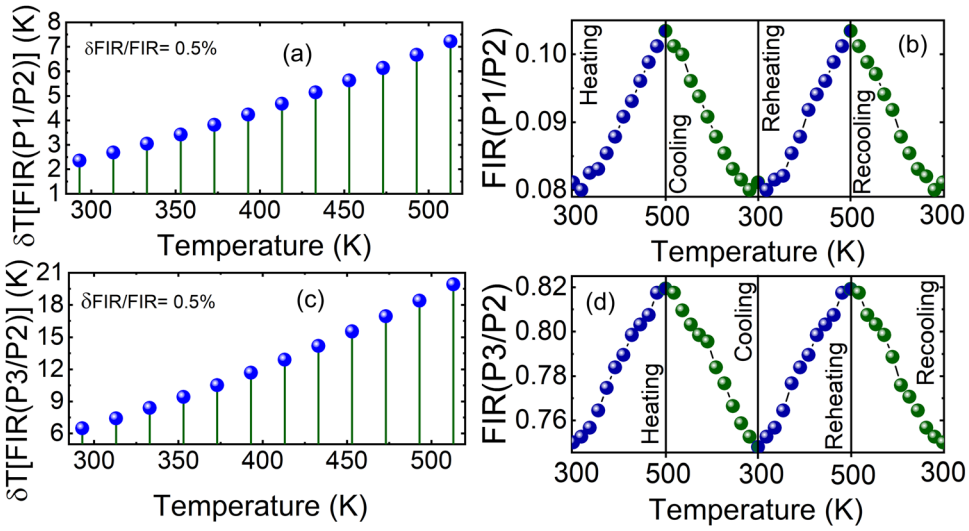


Fig. 12 (a) $\delta T$ (K) of FIR($P1/P2$), (b) FIR($P1/P2$) stability of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6% $Sm^{3+}$ for one cycle, (c) $\delta T$ (K) of FIR($P3/P2$), (d) FIR($P3/P2$) stability of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6% $Sm^{3+}$ phosphor for one cycle.

Therefore, the moderate sensitivity does not represent a significant drawback for integration into temperature sensing platforms, especially in environments where robustness and reliability are more critical than maximum sensitivity.

### 3.5. Chromaticity coordinates

It is widely recognized that the three fundamental colors perceived by humans are red, blue and green, commonly referred to as 1931 chromaticity coordinates. As a general rule, a color can be represented by a set of International Commission on Illumination (CIE) parameters for any light source. CIE parameters, including chromaticity coordinates ($x$, $y$) and CCT (Correlated Color Temperature), were computed to describe the color emission.[73]

Fig. 13(a) presents the CIE chromaticity diagram for $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$ phosphors at various $Sm^{3+}$ doping concentrations under 404 nm excitation. As observed, all the studied phosphors emit in the orange-red spectral region (see a real photograph of the color emitted in Fig. 13(b)), with CIE coordinates approximately constant at ($x$ = 0.60, $y$ = 0.39) across all doping concentrations. Notably, these calculated CIE coordinates match closely with the standard coordinates for orange-red emission. The minimal variation in chromaticity with increasing $Sm^{3+}$ concentration indicates that $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$ phosphors possess excellent color stability, making them promising candidates as orange-red emitting materials for solid-state lighting and semiconductor display applications. Chromaticity coordinates were calculated using Color Calculator software, and CCT was determined using McCamy's empirical formula:[74]

$$\mathrm{CCT} = -437n^3 + 3601n^2 - 6861n + 5514.31, \quad (18)$$

where $n = (x - x_e)/(y - y_e)$, and the epicenter of chromaticity is located at $x_e$ = 0.3320 and $y_e$ = 0.1858. CCT results calculated for

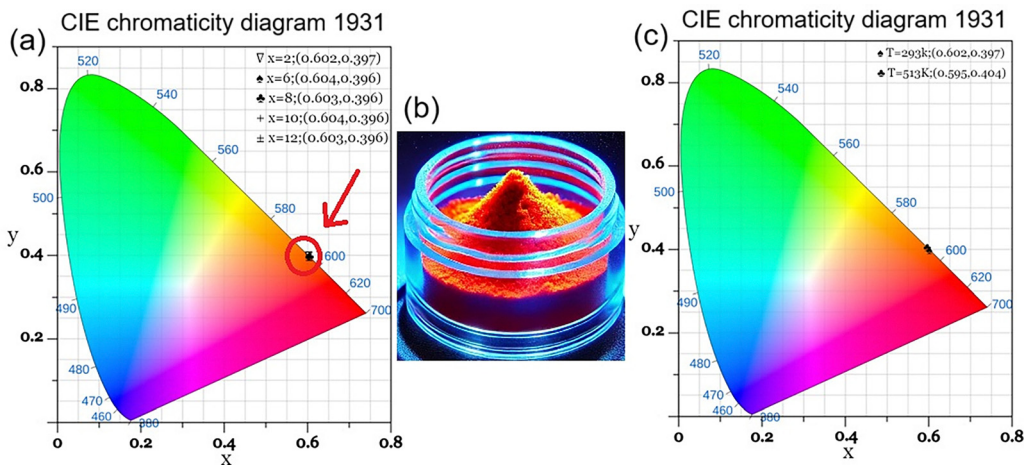


Fig. 13 (a) Chromaticity coordinates of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$ at% $Sm^{3+}$ ($x$ = 2, 6, 8, 10 and 12); (b) a photograph of the orange-red $Ca_3La_7(SiO_4)_5(PO_4)_5O_2$:6 at% $Sm^{3+}$ phosphor under blue light illuminant at room temperature; (c) chromaticity coordinates of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^3$ phosphors (at 293 K and 513 K) in the CIE 1931 chromaticity diagram.


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Table 3 CIE chromaticity coordinates ($x$, $y$), CCT values and color purities of synthesized samples $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$ at% $Sm^{3+}$ ($x$ = 2, 6, 8, 10 and 12)

| | CIE coordinates ($x$, $y$) | | | |
|---|---|---|---|---|
| $Sm^{3+}$ concentration (at%) | $x$ | $y$ | CCT (K) | Color purity (%) |
| 2 | 0.602 | 0.397 | 1717.49 | 96.20 |
| 6 | 0.604 | 0.396 | 1721.86 | 96.42 |
| 8 | 0.603 | 0.396 | 1720.10 | 95.53 |
| 10 | 0.604 | 0.398 | 1721.79 | 97.66 |
| 12 | 0.603 | 0.396 | 1721.47 | 95.31 |

$Ca_3La_7(SiO_4)_5(PO_4)O_2$:$x$$Sm^{3+}$ with $x$ = 2, 6, 8, 10 and 12 at% are reported in Table 3. The CCT values are all around 1700 K, which is less than 5000 K. These results verify that these phosphors can be successfully employed as near-UV-excited orange-red emitting phosphors.

Furthermore, the CIE coordinates of the $Ca_3La_7(SiO_4)_5(PO_4)O_2$:6 at% $Sm^{3+}$ sample underwent only minimal change during heating, rising only slightly from (0.602, 0.397) at 293 K to (0.595, 0.404) at 513 K (Fig. 13(c)). This high thermal stability, combined with excellent color purity, highlights the material's potential for high-performance applications. Color purity, which quantifies emission quality, can be calculated using the formula:[75]

$$\text{Color purity} = \frac{\sqrt{(x-x_i)^2+(y-y_i)^2}}{\sqrt{(x_d-x_i)^2+(y_d-y_i)^2}} \times 100\%, \qquad (19)$$

where ($x$, $y$) are the CIE coordinates of the phosphorescent sample under near-UV light irradiation (404 nm). ($x_i$, $y_i$) are the chromaticity coordinates of Illuminant $C$, which is a standard reference illuminant with a correlated color temperature of approximately 6774 K. In this study, the values of ($x_i$, $y_i$) are (0.31006, 0.31616). ($x_d$, $y_d$) are the chromaticity coordinates corresponding to the dominant wavelength of 598 nm, derived from the emission spectrum and calculated using CIE standard formulas. For this study, ($x_d$, $y_d$) = (0.618, 0.382).

The color purity of the synthesized samples can also be observed in Table 3 for various $Sm^{3+}$ concentrations. Notably, the color purity exceeds 95% across all doping concentrations, highlighting the phosphors' excellent orange-red emission quality. The orange-red emission of $Sm^{3+}$ phosphors complements other spectral components in WLEDs, addressing the red-light deficiency that commonly reduces the CRI in such systems. By contributing to a more balanced spectral distribution, these phosphors improve CRI, ensuring more accurate color reproduction. These findings, combined with stable CCT values around 1700 K, demonstrate the suitability of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$ phosphors for warm-white solid-state lighting applications that demand enhanced CRI alongside high color purity.

## 4. Conclusions

In summary, a series of $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$ were fabricated using a solid-state synthesis method, followed by extensive structural and luminescence evaluations. XRD analysis verified the phase purity of the synthesized materials, identifying a hexagonal crystal structure with the space group $P6_3/m$. In addition, Rietveld refinement provided confirmation of the successful incorporation of $Sm^{3+}$ ions into the crystal lattice. An analysis of the excitation spectra has confirmed that the phosphor material $Ca_3La_7(SiO_4)_5(PO_4)O_2$ doped with $Sm^{3+}$ ions can be efficiently activated using near-UV light. The emission spectra of these $Ca_3La_7(SiO_4)_5(PO_4)O_2$:$Sm^{3+}$ phosphors display prominent orange-red luminescence. This emission is influenced by multipole–multipole energy transfer mechanisms within the phosphor structure, contributing to its high luminescence intensity. The optimal concentration of $Sm^{3+}$ in $Ca_3La_7(SiO_4)_5(PO_4)O_2$ matrix was determined by experimentation to be 6 at% for achieving maximum luminescence efficiency. The exchange interaction between $Sm^{3+}$–$Sm^{3+}$ ions being responsible for the concentration quenching that occurs at a critical distance of 26 Å. Furthermore, the phosphor demonstrated excellent color purity, reaching 97.66%, with CIE chromaticity coordinates of (0.604, 0.398) and a low correlated color temperature of approximately 1721 K. The decay time for $Sm^{3+}$-doped phosphors was approximately 1 ms for samples with 6 at% $Sm^{3+}$, a moderate response time suitable for various LED applications. Additionally, the relative sensitivity of $Ca_3La_7(SiO_4)_5(PO_4)O_2$ doped with 6 at% $Sm^{3+}$ was measured to be around 0.212% $K^{-1}$ at 293 K, highlighting the remarkable thermal sensitivity of these $Sm^{3+}$-doped samples. In short, this work presents innovative $Sm^{3+}$-doped oxyapatites with promising applications in both solid-state lighting technologies and thermal sensors.

## Data availability

The datasets generated and analyzed during this study are not publicly available to ensure compliance with institutional policies and potential confidentiality agreements. However, they can be made available upon reasonable request. Researchers interested in accessing the data should contact the corresponding author, Dr Xavier Mateos, at xavier.mateos@urv.cat. Requests should include a brief description of the intended use of the data and any relevant research context.

## Conflicts of interest

There are no conflicts of interest to declare.

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