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Ozonolysis–oxidation-driven top-down strategy for the target preparation of ultrathin 2D metal–organic framework monolayers†

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Two-dimensional metal–organic–framework (2D MOF) nanosheets with a modular nature and tunable structures exhibit a bright future for sensors, separation, and catalysis. Developing ultrathin 2D MOF nanosheets with unique physical and chemical properties is urgently required but very challenging. Although the chemical exfoliation strategy has been regarded as a promising way to exfoliate large amounts of three-dimensional (3D) pillar-layered MOFs, this method still faces many problems and remains insufficient. In this study, a novel chemical exfoliation method is developed for the target preparation of 2D MOF monolayers from the 3D pillar-layered MOFs. The Co/Zn/Cu–MOFs with a pillar ligand of *trans*-1,2-bis(4-pyridyl)ethylene (bipyen) are subjected to be broken by the cleavage of C=C bonds in the bipyen ligands *via* an ozone oxidation reaction. As chemical exfoliation is processed *via* the oxidation of the pillar ligand by ozone, the thickness of the 2D MOFs can be tuned by the control of oxidation time and the obtained 2D Co/Zn/Cu–MOF monolayers are functionalized with a –COOH group. This study provides an effective and general chemical exfoliation method to prepare monolayer MOFs from the 3D pillar-layered MOFs with bipyen as the pillar ligand.

Keywords: 3D pillar-layered MOFs; Ultrathin 2D MOF monolayers; Top-down strategy; Chemical exfoliation; Ozonolysis–oxidation.

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1 Introduction

Two-dimensional (2D) materials with atomic or molecular thickness levels have attracted considerable attention owing to their unique physical and chemical properties in many areas, such as electronics, sensors, biomedicine, catalysis, separation, energy conversion, and storage.^{1–7} As an emerging important family of multifunctional 2D materials, 2D metal–organic–framework (2D MOF) nanosheets with a modular nature and tunable structures, including structural diversity, large specific surface area, high porosity, and controllable pore size, have a bright future for many applications, such as

sensors, separation, and catalysis.^{8–13} These attractive features and promising applications have endowed 2D MOF nanosheets as a new competitive member of 2D nanomaterials. Therefore, the precise synthesis of ultrathin 2D MOF nanosheets with controllable multiple aspect ratios and integrated phase structure has received intensive interest in the past years.^{11,14–17}

In general, two strategies have been developed for the synthesis of 2D MOF nanosheets.^{18–20} One is the bottom-up approach, by which the MOF nanosheets are directly and controllably assembled by the corresponding metal ions and organic linkers in solution.^{21–23} However, the bottom-up approach is largely limited by the reaction conditions, which cannot be applied as a universal method. The other one is the top-down approach, by which 2D MOF nanosheets are obtained *via* physical or chemical exfoliation of the bulk MOF.^{24–26} Compared with the bottom-up synthesis strategy, the top-down method is more widely applicable. For example, the way of physical exfoliation has been regarded as the most powerful and scalable approach for the exfoliation of almost all 2D layered MOFs through exogenous physical force or solvent mediation to regulate the interlayer interaction.^{27–30} While the chemical exfoliation strategy has been regarded as

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Fig. 2 (a) SEM image and (b) experimental and simulated XRD patterns of the 3D pillar-layered Co-MOF.

images reveal that the 3D pillar-layered Co-MOF has a block shape morphology with conspicuous multilayer structure (Fig. 2a and S2, ESI[†]). PXRD data confirms phase purity of the prepared 3D Co-MOF precursor, which exhibit the same diffraction peaks as those of simulated pattern (Fig. 2b). The diffraction peak located at $2\theta = 5.5^\circ$ can be attributed to the (001) crystal plane of the 3D Co-MOFs, corresponding to the crystal face composed of the pillar ligand biphenyl.

TEM images indicate that the Co-MOF precursor still maintains its 3D pillar-layered structure after ozone treatment for 30 min (Fig. 3a). As the reaction duration increases to 60 min, 2D Co-MOF nanosheets are observed (Fig. 3b). Moreover, thinner 2D Co-MOF nanosheets are obtained while the time prolongs from 60 min to 90 min (Fig. 3c), displaying the successful exfoliation of 3D pillar-layered Co-MOF. It is observed that the obtained nanosheets exhibit wrinkled and curved morphologies, showing the characteristics of ultrathin thickness. In addition, some small sized nanosheets are adsorbed on the 2D Co-MOF nanosheet due to the electrostatic adsorption. Finally, the 2D Co-MOF nanosheets exhibit low lateral dimension and even aggregated together to form nanoparticles while the duration time prolongs to 120 min (Fig. 3d). This may be attributed to

the destruction of the layered structure by the excessive ozone. These identifications are further approved by the XRD patterns of exfoliated Co-MOFs under different reaction duration. It is observed that the intensity of the diffraction peak at 5.5° gradually decreased and finally disappeared as the ozone treatment time going by (Fig. 3e). This indicates the loss of (001) crystal plane while the C=C bonds in pillar ligand biphenyl are broken. As shown in the atomic force microscope (AFM) data, the thickness of the 2D Co-MOF nanosheet (1.6 nm, Fig. 3g) is close to the simulation value of the theoretical single-layer crystal (Scheme 1). These results show that the chemical exfoliation process of this 3D pillar-layered MOF can be fine-tuned by the control of ozone treatment duration.

For the chemical exfoliation mechanism, the formed 2D Co-MOF nanosheets are expected to functionalize with -COOH groups on their surface. As shown in Fig. 4a, the 2D Co-MOF nanosheet exhibits a negative zeta potential of -25 mV than that of its 3D Co-MOF precursor (-4.5 mV), indicating more negative charge feature due to the dissociation of the -COOH groups. To verify the presence of -COOH groups, the 3D Co-MOF and 2D Co-MOF nanosheets are dissolved in DCl, and solvents containing the organic



Fig. 3 TEM images of the 3D Co-MOF subjected to a chemical exfoliation for different reaction durations of (a) 30 min, (b) 60 min, (c) 90 min, and (d) 120 min; (e) XRD patterns of the 3D Co-MOF subjected to a chemical exfoliation for different reaction durations; (f) AFM image of the 3D Co-MOF being exfoliated for 90 min; and (g) the corresponding thickness of the resulted nanosheets.





Fig. 4 (a) Zeta potential of the 3D Co-MOF precursor and 2D Co-MOF nanosheets; (b) the ^{13}C NMR spectra of the solvents derived from 3D Co-MOF precursor and 2D Co-MOF nanosheet.

ligands are measured by the ^{13}C NMR spectroscopy. It is seen from Fig. 4b that the chemical shifts at 125.3, 129.9, 142.7, and 149.6 ppm correspond to the pillar ligand of bipyen, while the chemical shifts at 125.4, 137.5, 150.0, and 165.6 ppm correspond to the isonicotinic acid. These results suggest the functionalization of the $-\text{COOH}$ groups on the 2D Co-MOF nanosheet.

To test the generality of this method, two similar 3D pillar-layered MOFs (Zn-MOF and Cu-MOF, as shown in Fig. S3, ESI †) with bipyen as a pillar ligand were prepared for the chemical exfoliation. The corresponding XRD patterns of the 3D Zn-MOF and 3D Cu-MOF further confirm the crystal structures (Fig. S4, ESI †). It can be seen from Fig. 5a and b that the 3D Zn-MOF and 3D Cu-MOF possess block shape

morphology with a conspicuous multilayer structure. 2D nanosheet morphology is observed in Fig. 5c and d after the chemical exfoliation for 90 min, confirming the successful chemical exfoliation. The EDS mapping results show that the 2D Zn-MOF and 2D Cu-MOF are composed of C, H, O, and Zn or Cu (Fig. S5 †), and the 2D nanosheet structures were confirmed by comparing with the simulated XRD patterns of the 2D Zn-MOF and 2D Cu-MOF (Fig. S6, ESI †). It was found that the 2D Zn-MOF has a thickness of 1.6 nm and the 2D Cu-MOF has a thickness of around 2.0 nm (Fig. 6), displaying their monolayer feature. Moreover, the obtained 2D Zn-MOF and 2D Cu-MOF nanosheets also exhibit negative zeta potentials than that of the 3D Zn-MOF and 3D Cu-MOF (Fig. S7, ESI †), indicating the functionalization of $-\text{COOH}$ groups.



Fig. 5 SEM images of the 3D pillar-layered (a) Zn-MOF precursor and (b) Cu-MOF precursor and TEM images of the (c) 2D Zn-MOF nanosheet and (d) 2D Cu-MOF nanosheet.





Fig. 6 AFM images and the corresponding thickness of the (a) and (b) 2D Zn-MOF nanosheet and (c) and (d) 2D Cu-MOF nanosheet.

3 Conclusions

In summary, the monolayer 2D Co/Zn/Cu-MOFs nanosheets functionalized with $-\text{COOH}$ groups were exfoliated from 3D pillar-layered MOFs by the convenient chemical exfoliation method. The 3D pillar-layered MOFs were easy to flake off *via* cleaving the covalent bonds of the pillar ligands by ozonolysis oxidation. Its chemical process can be controlled by the oxidation duration of the pillar ligand with ozone. This study provides an effective and generality chemical exfoliation method to prepare monolayer MOFs from the 3D pillar-layered MOFs with bipyen as a pillar ligand.

4 Experimental section

4.1 Chemicals

$\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (AR, >99.0%, Aladdin), $\text{Zn}(\text{NO}_3)_2 \cdot 2\text{H}_2\text{O}$ (AR, >99.0%, Guangdong Guanghua Chemical Factory Co., Ltd.), $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$ (AR, >99.0%, Guangdong Guanghua Chemical Factory Co., Ltd.), *trans*-1,2-bis(4-pyridyl)ethylene (Bipyen, 98%, Aladdin), *trans*-1,4-cyclohexanedicarboxylic acid (H_2CDC , 97%, Aladdin), 2,6-naphthalenedicarboxylic acid (H_2NDC , 98%, Energy Chemical), acetone (AR, >99.0%, Guangdong Guanghua Chemical Factory Co., Ltd.), *N,N*-dimethylformamide (DMF, AR, >99.5%, Guangdong Guanghua Sci-Tech Co., Ltd.), ethanol (AR, >99.5%, Guangdong Guanghua Sci-Tech Co., Ltd.), hexane (AR, >99.5%, Guangdong Guanghua Sci-Tech Co., Ltd.), and potassium iodide (AR, >99.0%, Aladdin). All the chemicals

were obtained from commercial sources and used as received.

4.2 Materials preparation

Preparation of 3D pillar-layered MOFs of $[\text{Co}_3(\text{NDC})_3(\text{bipyen})_{1.5}] \cdot \text{H}_2\text{O}$ (3D Co-MOF). The synthesis of the 3D Co-MOF was carried out according to the previous report.³⁵ Typically, 87.3 mg $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (0.3 mmol), 64.9 mg H_2NDC (0.3 mmol), 54.7 mg bipyen (0.3 mmol) were dissolved in 15 mL H_2O and reacted in a 25 mL Teflon reactor at 180 °C for 48 h. The dark green product was collected by filtration and washed with H_2O , ethanol, and acetone, then vacuum-dried at 70 °C overnight. The productivity was only 5%.

Preparation of 3D pillar-layered MOFs of $\text{Zn}(\text{NDC})(\text{bipyen})_{0.5} \cdot 2.5\text{DMF} \cdot 0.5\text{H}_2\text{O}$ (3D Zn-MOF). The synthesis of the 3D Zn-MOF was carried out according to the previous report.³⁹ Generally, 331.4 mg $\text{Zn}(\text{NO}_3)_2 \cdot 2\text{H}_2\text{O}$ (1.47 mmol), 317.8 mg H_2NDC (1.47 mmol), and 134.8 mg bipyen (0.74 mmol) were dissolved in 60 mL DMF and reacted in a 150 mL distillation bottle at 100 °C for 24 h. The colorless product was collected by filtration and washed with DMF and hexane, then vacuum-dried at 70 °C overnight. The productivity is 75%.

Preparation of 3D pillar-layered MOFs of $\text{Cu}_4(\text{CDC})_4(\text{bipyen})$ (3D Cu-MOF). The synthesis of the 3D Cu-MOF was carried out according to the previous report.⁴⁰ Generally, 56.3 mg $\text{Cu}(\text{NO}_3)_2 \cdot 2.5\text{H}_2\text{O}$ (0.242 mmol), 63.0 mg H_2CDC (0.366 mmol), and 22.1 mg bipyen (0.121 mmol) were



dissolved in 15 mL H₂O and reacted in a 25 mL Teflon reactor at 80 °C for 72 h. The blue-green product was collected by filtration and washed with H₂O, ethanol, and acetone, then vacuum-dried at 70 °C overnight. The productivity is only 10%.

Preparation of 2D MOF nanosheets. The 3D pillar-layered MOFs were first ball milling to obtain a uniform size thus facilitate subsequent chemical exfoliation process. The 3D pillar-layered MOFs were still remaining the pillar-layered structure after ball milling step. The ball milling process was as follows: 50 mg 3D MOFs and 15.0 mL ethanol were added into the agate jar and ball milling at 400 rpm min⁻¹ for 2 h. Then, the obtained dispersion was filtered and vacuum-dried at 70 °C overnight. In a typical ozone oxidation experiment, 15 mg of the above ball-milled 3D MOF was dispersed in 20 mL DMF under sonication in a 50 mL two-necked round bottom flask and then cooled to 0 °C. Then, the ozone gas flux (5.28 mmol h⁻¹) was passed into the mixture for 60 min, 90 min, and 120 min under sonication (100 W, 60 kHz), and the exhaust gas was absorbed by potassium iodide solution. The exfoliated ultrathin 2D MOF nanosheets were collected from the upper colloidal suspension after sedimentation for 2 h by centrifugal separation (12 000 rpm min⁻¹ for 15 min) and then dispersed in ethanol for freeze-drying.

4.3 Physicochemical characterization

PXRD were carried out on an X-ray diffractometer (Cu K α radiation, 40 kV, 40 mA, D8 Advance, Bruker, Germany) at a scan rate of 0.02° per second in the 2 θ range from 5° to 30°. The morphology and microstructures were characterized by scanning electron microscopy (SEM, SU8220, Japan) under an acceleration voltage of 10.0 kV and transmission electron microscopy (TEM, JEM-2100HR, JEOL, Japan) at an accelerating voltage of 200 kV. Atomic force microscopy (AFM) images were obtained by a Bruker Multimode 8 device. NMR spectra were recorded on Bruker-500 (500 MHz for ¹H, 500 MHz for ¹³C) instruments internally referenced to the SiMe₄ signal. Zeta potential was performed by zeta-potential tester (90 Plus, Brookhaven Instruments Corporation, America).

4.4 Ozone treatment of *trans*-1,2-bis(4-pyridyl)ethylene

First, 10 mg *trans*-1,2-bis(4-pyridyl)ethylene was dissolved in 20 mL DMF in a 50 mL two-necked round bottom flask. Second, the ozone gas (5.28 mmol h⁻¹) was bubbled into the solution for 40 min under sonication in an ice-water bath. Then, the product was collected by evaporating the DMF and analyzed by ¹H NMR.

Conflicts of interest

The authors declare no conflict of interest.

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