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## Strategies for interface issues and challenges of neural electrodes†

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Neural electrodes, as a bridge for bidirectional communication between the body and external devices, are crucial means for detecting and controlling nerve activity. The electrodes play a vital role in monitoring the state of neural systems or influencing it to treat disease or restore functions. To achieve high-resolution, safe and long-term stable nerve recording and stimulation, a neural electrode with excellent electrochemical performance (e.g., impedance, charge storage capacity, charge injection limit), and good biocompatibility and stability is required. Here, the charge transfer process in the tissues, the electrode–tissue interfaces and the electrode materials are discussed respectively. Subsequently, the latest research methods and strategies for improving the electrochemical performance and biocompatibility of neural electrodes are reviewed. Finally, the challenges in the development of neural electrodes are proposed. It is expected that the development of neural electrodes will offer new opportunities for the evolution of neural prosthesis, bioelectronic medicine, brain science, and so on.

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### 1. Introduction

Since Luigi Galvani discovered bioelectricity,<sup>1,2</sup> it has given us an effective way to understand the behaviors of biological tissues by electrophysiological monitoring *via* external electronic devices, the so-called bioelectronic interface or bioelectronics.<sup>3</sup> Bioelectronics is an emerging subject constituted by the mutual penetration of biology and electronic information science. Almost every physiological process in the human body



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is related to bioelectricity. For instance, heart beats, muscle contraction, brain activities, and so on.<sup>4</sup> The neural electrodes, which help to exchange information between the biological tissues and external electronic devices in a bidirectional way (stimulus and recording),<sup>5</sup> have preeminent potential in the treatment of various neurological diseases (*e.g.*, depression, nerve palsy, epilepsy, Alzheimer's disease, blindness and spinal cord injury) and the exploration of how the brain works.<sup>6–11</sup>

The interface between the biological tissue and the electrode is a major element that affects the performance of neural electrodes.<sup>12</sup> Therefore, building a dependable and efficient nerve-electrode interface is crucial for the development of neural electrodes.<sup>13,14</sup> To achieve a high performance nerve-electrode interface, attention should be paid to the following aspects: (i) the charge transfer mechanism of diverse electrode materials needs to be clarified, since the electrical deviations of neural interfaces bring about obstacles to signal transmission and detection.<sup>15–17</sup> (ii) An excellent electrochemical performance of the neural electrode is highly desired,<sup>18,19</sup> including electrochemical impedance,<sup>20,21</sup> charge storage capacity (CSC),<sup>22,23</sup> charge injection limit (CIL, it is defined as the maximum charge density that can be injected into the tissue under the safe potential window measured by cyclic voltammetry)<sup>24–26</sup> and so on. Among them, a low impedance favors monitoring the electrophysiological signal with more details. A high charge storage capacity and charge injection limit enhance the stimulating capability of the electrode (Table S1†). In addition, the flexible stretchable/bendable<sup>27,28</sup> electrodes can withstand large mechanical deformations and conformal well with soft dynamic biological tissues, which improves the fidelity and stability of the signal transmission. Simultaneously, high-density neural electrodes can achieve multi-site acquisition of the same neuron activity,<sup>29,30</sup> so that more waveform information can be used to distinguish neuron signals from multiple sources to ensure the authenticity of the information. (iii) Until now, the weak biocompatibility of the neural electrodes is still a rigorous problem.<sup>22,31,32</sup> Although helpful tissue adhesion and survival of nerve cells were observed *in vitro*, diverse degrees of tissue reaction occurred *in vivo*, which weakens the transmission of signals of the electrodes. Deservedly, it is imperative to comprehend and resolve the trouble of poor compatibility, which hinders the way for long term and stable implantation of the neural electrodes.

To further promote efficacious and credible information exchange between electronic devices and biological tissues, it is necessary to summarize the works that focused on addressing the above issues. Accordingly, in this review, we firstly discussed the feature of the neural electrode interface, the signal transmission mechanism between biological tissue and the neural electrode, and proposed countermeasures for the design of the electrode–tissue interface. Subsequently, the development of the neural electrodes is summarized and discussed according to the charge transfer mechanism of the electrode materials, including (i) charge transfer by electrons, (ii) charge transfer by ions, and (iii) charge transfer *via* a syner-

getic effect of electrons and ions. Finally, we analyse and summarize the challenges encountered in the practical application of next-generation neural electrodes, hoping that this paper will bring new insights for the seamless integration of biology and electronics.

## 2. Neural electrode–tissue interfaces

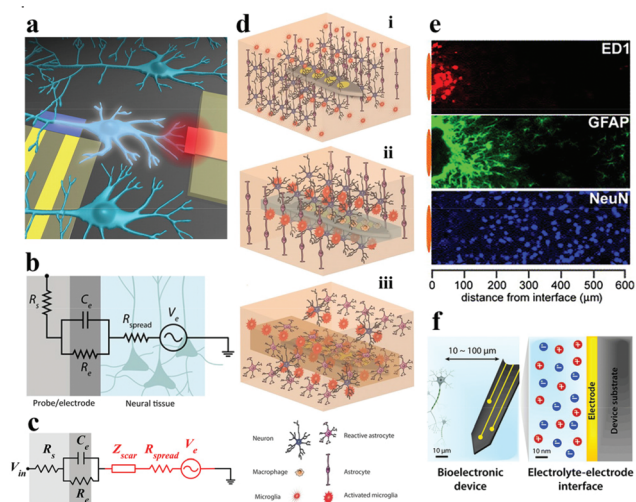
At present, intelligent robots with mechanical bodies and emulating human intelligence already exist. However, in order to truly assist or even replace human beings to complete dangerous, arduous and intricate work, and to serve human life with high efficiency and perfect quality, one of the vital issues that needs to be settled is conspicuous signal transmission and mechanical control between brain and machine. In this procedure, the interface between the machine and the body is the most crucial part. Currently, flexible electronic devices have become an innovative solution for the information communication of the machine–body interface after being tightly integrated with tissues.

The research of the neural interface is a multidisciplinary investigation field, which relies on the combination of multiple disciplines, including materials science, mechanics, electronics, medicine and so on. The neural interface refers to the interface between a biological tissue and an electrode constructed to detect physiological signals or emit electrical stimulation to the skin or an inside organism.<sup>17,33,34</sup> Markedly, the execution of some functions of the interface (the transmission of stimulation and recording signals) requires the electrode to possess good biocompatibility, low electrochemical impedance, long term stability, excellent electrical conductivity and so on.

Due to the different charge transfer behaviour of neural electrodes and biological tissues/cells, where the charge transfer in the electrode materials mainly relays on electrons and the tissues/cells use ions to transfer charge, an effective and stable information transfer of the neural interface is important for the construction of neural electrodes (Fig. 1a).<sup>35,36</sup> The optimization of mechanical interaction and understanding of the charge transfer mechanism between the electrode and tissues could favour the establishment of an effective neural interface, so as to promote the development of neural electrodes. Therefore, in the consequent part, we discussed the mechanical interaction between the electrode and tissue. Then, we further discussed the signal transmission mechanism in electrode materials, tissues and electrode–tissue interfaces.

### 2.1. Mechanical interaction of the electrode–tissue interface

Favorable biomechanical interactions require the electrode to maintain splendid parameters (*e.g.*, biocompatibility, stability, electrochemical performance, *etc.*). The biomechanical interactions of nerve electrodes are generally categorized into two types: one is the noninvasive electrode with epidermal attachment, and the other one is the invasive electrode with *in vivo*



**Fig. 1** (a) Schematic of the gap between bioelectronic interfaces. Reproduced with permission from ref. 37. Copyright 2016, the American Chemical Society. (b) Illustration of the equivalent circuit between the electrode–tissue interface, (c) effect of the neuroinflammatory response ( $Z_{\text{scar}}$ ) on partial electrical impedance. Reproduced with permission from ref. 20. Copyright 2017, American Association for the Advancement of Science. (d) The process of glial encapsulation after electrode implantation: illustration of (i) implanted nerve electrode, and (ii) acute and (iii) chronic responses of the electrode. Adapted with permission from ref. 38. Copyright 2014, the Wiley-VCH. (e) The foreign body reaction in the brain tissue of animals implanted with microelectrodes for 4 weeks may be caused by the formation of inflammatory cells (ED1), astrocytes (GFAP) and the increased distance (NeuN) of neurons from the recording position. Reproduced with permission from ref. 39. Copyright 2005, the Elsevier. (f) Schematic description of signal conversion between the electrode and target cell interface in biological systems usually occurs at 10–100  $\mu\text{m}$ . Reproduced with permission from ref. 21. Copyright 2019, the Royal Society of Chemistry.

implantation. Epidermal electrodes universally trigger less invasive biomechanical interactions when in conformal contact with soft skin tissue (*e.g.*, curving, stretching, compression, *etc.*).<sup>40</sup> However, more complex biomechanical interactions (*e.g.*, puncture wounds and destruction of the microvascular system) occur with invasive electrodes, especially the implantable nerve electrodes.<sup>21</sup>

The electrode is taken as an example for recording the electroencephalogram (EEG). With the aggrandizement of time, the blood–brain barrier will be destroyed and blood vessels will be damaged, resulting in the death of a large number of nerve cells around the electrode (Fig. 1d).<sup>41</sup> At this point, the immune system of the body is activated. Microglia and macrophage cells (derived from blood) are activated and release chemokines that attract astrocytes towards the electrode surface to aggregate (Fig. 1e). Astrocytes secrete some neurotoxic factors and the extracellular matrix, which lead to the death of neurons around the electrodes and the formation of dense glial scars, respectively.<sup>42</sup> The glial scar will fortify the distance between the electrode and the target neuron, and at the same time, the impedance of the neural interface will increase dramatically. Equivalent circuit models can be

employed to intuitively describe the impedance increase at the electrode–tissue interface (Fig. 1b and c). Higher interface impedance will seriously impede the acquisition of bioelectronic stimulation and recording signals (Fig. 1f).<sup>20,37,43,44</sup> When the glial scar becomes too bushy and the neurons around the electrode die due to inflammation (acute and chronic),<sup>45,46</sup> the electrode loses its function. So far, methods for perfecting the compatibility of bioelectrodes include: (i) surface coating;<sup>47</sup> (ii) doping;<sup>48</sup> (iii) covalent grafting;<sup>49</sup> and (iv) layer-by-layer self-assembly technology based on electrostatic attraction.<sup>50</sup>

Besides the reasons described above that cause the increase of impedance, it is discovered that equipment/electrode degradation induced by operation in a biological environment also could result in an impedance increase. Therefore, improving the electrochemical performance and enhancing the biocompatibility of the neural electrode are the most important requirements for its practical applications. These requirements could be satisfied to a certain extent by the following strategies: (i) minimizing the gap of the neural interfaces;<sup>51</sup> (ii) improving the interface adhesion between the electrode and the tissue;<sup>52</sup> (iii) reducing the electrode thickness;<sup>40</sup> and (iv) enabling the electrode with porous/3D structures.<sup>53</sup>

## 2.2. Signal transmission between the neural electrode and tissue

It is an important parameter to evaluate the electrode–tissue interface that whether the electrode can monitor the electrophysiological signals from the nervous system and accurately stimulate them.<sup>54</sup> Hence, in this part, we would like to discuss the charge transferring mechanism in the neurons, electrode–tissue interface, and electrode materials, so that to provide theoretical support and design guidelines for the precise fabrication of neural electrodes. Notably, this article only focuses on the electrical interface directly formed between the electrode and the nerve tissue. The discussion of the chemical and optical modes of the neural interface could be found in previously reported reviews.<sup>5,55</sup>

### 2.2.1. The transmission of electrophysiological signals between neurons.

The nervous system completes the transmission of information through the interaction of electrical carriers (ions) and chemical carriers (neurotransmitters).<sup>56</sup> In nerve cells (neurons), information is transmitted along the nerve fibers in the form of electrical signals (Fig. 2). When the nerve fibers are stimulated and become exciting, the membrane potential transforms from internally negative and externally positive ( $\text{K}^+$  external flow, both intracellular  $\text{K}^+$  and extracellular  $\text{Na}^+$  are excessive in the resting state) to internally positive and externally negative ( $\text{Na}^+$  internal flow), and thus the potential difference is generated, leading to charge movement. Finally, the excitement is transmitted in the form of a local current signal. Consequently, the transmission of information from one neuron to another neuron or target organ is realized through chemical synapses.<sup>55</sup> The process begins with the transmission of excitement to the axon and then to the synapse, where the synaptic vesicles move forward after being stimulated. Depolarization occurs when the action potential



**Fig. 2** The progressively amplified communication diagram from a neuron to a synapse to an ion channel. (a) neurons communicate through chemical and electrical signals, (b) the presynaptic membrane releases neurotransmitters, which then spread *via* the synaptic cleft and integrate to receptor proteins on the postsynaptic membrane, (c) multi-formity of neuronal membrane receptors included in neurotransmission. Reproduced with permission from ref. 5. Copyright 2019, Springer Nature.

reaches the presynaptic membrane (composition of the axon membrane), and voltage-gated  $\text{Ca}^{2+}$  channels at the presynaptic terminal are opened to allow  $\text{Ca}^{2+}$  to flow into the cell, where intracellular calcium-mediated vesicles fuse with the presynaptic membrane and release neurotransmitters.<sup>57</sup>

This neurotransmitter reaches the postsynaptic membrane (constituted by a dendritic membrane or cell body membrane) by chemical diffusion and binds to a specific receptor molecule on the membrane, causing the channel to be opened and  $\text{Na}^+$  to enter the postsynaptic cells. When the membrane potential threshold exceeds the unit's potential, the action potential is triggered, and the neuron signal is transmitted from the presynaptic membrane to the postsynaptic membrane, which makes another neuron excited or inhibited.<sup>58</sup> Subsequently, the neurotransmitter is reabsorbed into the presynaptic cells, which completes the transmission of synaptic signals. In brief, the electrical communication between neuron cells mainly depends on the flow of ions.

**2.2.2. Charge transfer at the electrode–tissue interface.** The transmission of signals from the nervous system is achieved by inducing ions to pass through the nerve cell membrane, which will change the concentration of ions in the extracellular fluid. These ions act as charge carriers in the extracellular fluid, and their movement will facilitate a shift of the electric field. Neural electrodes communicate with biological tissues in a bidirectional mode, where one is to record the changes in the electric field of the tissues, and the other is to deliver electricity to alter the electric field near the tissues to stimulate or inhibit them.<sup>59,60</sup> No matter which mode of communication is adopted at the neural interface, it is important to hire electroactive materials to guarantee the efficient cross-interface charge transfer between ions and electrons, since the charge in the nerve electrode is carried by the electrons in the conductor, while it is carried by ions in the tissues.<sup>36,55</sup> Generally, the diverse charge transfer processes at the electrode/tissue interface can be categorized as two primary types, as demonstrated in Fig. 3.



**Fig. 3** (a) Schematic drawing of the capacitive charge injection mechanism (electric double layer), (b) Faraday charge injection mechanism (electrochemical process), and (c and d) their corresponding cyclic response.

In one way, the charge transfer depends on the charging and discharging process of the electric double-layer capacitor formed on the electrode surface (Fig. 3a and c). When the electrode transmits an electric pulse, the concentration of electrostatic charge on the surface will change, accompanied by alternate attraction and exclusion of the ions in the tissue fluid. There is no electron transfer between the electrode and the electrolyte during this process. More meaningfully, the electrode adsorbs a layer of polarized water molecules on the surface to act as a dielectric for the electric double layer capacitor.<sup>61</sup> Notably, the capacitive charge transfer mechanism will not produce or consume chemicals, so it is more appropriate for the practical application of neural electrodes. Nevertheless, the capability of charge transfer of the electrode depends on the capacitance of the electric double layer capacitor, and the capacitance is positively related to the electrode area. However, normally, the electrode areas are too small to satisfy the requirement of high-density charge injection. Enlarging the surface area of the neural electrode will promote its development and application in the neural interface.

In the other way, the charge transfer relies on the electrochemical reaction on the electrode surface, which is a Faraday procedure (Fig. 3b and d). That is to say, when the electrode transmits an electrical impulse, the electrode–tissue interface will correspondingly undergo a redox reaction. The procedure is accompanied by the oxidation or reduction of chemical substances, which manifests as electrons pass through the elec-

trode–tissue interface. The Faraday charge transfer process is divided into irreversible and reversible routes according to whether new substances are eventually generated. In the irreversible Faraday process, an irreversible oxidation–reduction reaction occurs, which will not only lead to electrode decay and variation of the pH value in tissue fluid, but also generate some harmful substances. Thus, it is better to avoid the charge transfer engaging through an irreversible Faraday pathway. In addition, for the reversible Faraday process, although the new material is formed on the electrode surface, when the opposite pulse current passes through the electrode, new substances will be absolutely reduced to the initial state. Therefore, the process does not import new materials to the tissue or bulk solution. Hence, the charge transfer *via* a reversible Faraday process is secure and feasible. It is vital that the corresponding redox reaction appears when the charge is injected in the Faraday reversible procedure, implying that the electrode interface can accommodate more charges. Consequently, the neural electrodes with a reversible Faraday charge injection process are preferred compared with the one with the irreversible Faraday process.

**2.2.3. Charge transfer in the electrode materials.** Effective and stable information transmission of the neural interface is challenging due to the mismatching of charge carriers at the electrode–tissue interface. We have discussed the mechanism of charge transfer among neurons (2.2.1) and at the electrode–tissue interface (2.2.2). Subsequently, we will summarize and discuss the charge transfer in the electrode materials. We categorized the charge transfer in the electrode materials into three types (Fig. 4): (i) Charge transfer by electrons: traditional neural electrode materials such as metal and carbon employ free electrons as movable charge carriers to communicate with biological tissues.<sup>55,62</sup> Because of their outstanding electrical conductivity and biological stability, they can be implanted *in vivo* for a long time and have been extensively investigated;<sup>63</sup> (ii) charge transfer by ions: hydrogels are essentially ionic conductive materials, which reveal favorable promise in neural interfaces due to their inherent biocompatibility, flexibility,

and compliance with nerve tissues;<sup>11,21,64–66</sup> and (iii) charge transfer *via* an electron–ion synergistic effect: the electrode materials include not only conductive polymers with soft, flexible and mechanically adjustable properties,<sup>67,68</sup> but also the composite materials which are constituted with both ion-conductive and electron-conductive materials.

### 3. Current developing status of neural electrodes

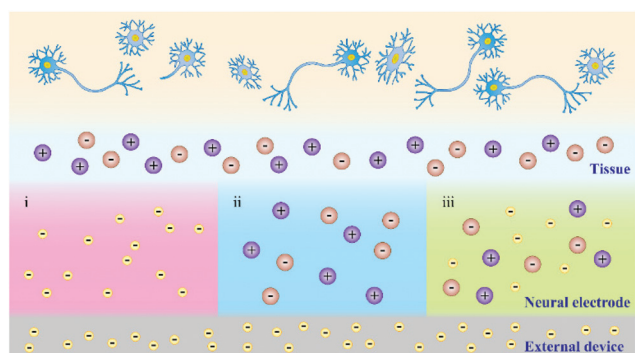
Considering the complicity and diversity of the neural interface, it is critical and challenging to select applicable materials to enable neural electrodes with high resolution, perfect electrochemical performance, excellent biocompatibility and long-term stability. In the following parts, we discuss the design schemes and performance evaluation of the reported neural electrodes in terms of their charge transfer mechanisms (Table S2†).

#### 3.1. Neural electrodes that transfer charge by electrons

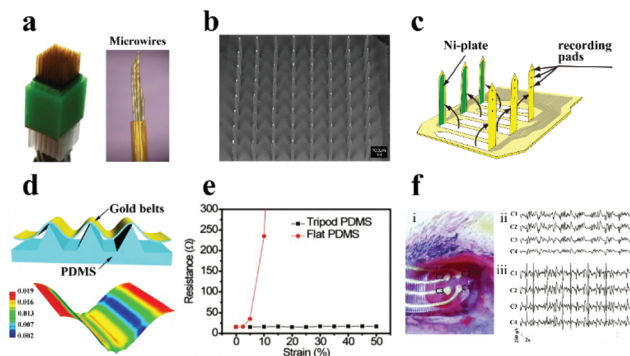
Innovative breakthroughs have been made in materials, sizes, shapes and textures of nerve electrodes to optimize their electrical properties, biocompatibility and mechanical properties. The neural electrodes made from metals or carbon materials, employ free electrons as mobile charge carriers to communicate with biological tissues.<sup>25,62</sup> Because of their outstanding electrical conductivity ( $10^6$ – $10^7$  S m<sup>-1</sup>),<sup>69</sup> long-term longevity, *etc.*,<sup>63</sup> they are broadly used in deep brain stimulators, cardiac pacemakers, retinas<sup>70</sup> and so on. In this chapter, we chiefly review the design strategies of neural electrodes that transfer charge by electrons.

**3.1.1. Metal electrode.** To date, the most commonly implemented neural electrodes are principally made of electron conductive materials, namely metals and metal composites (*e.g.*, platinum,<sup>71</sup> iridium,<sup>72</sup> and platinum–iridium<sup>73</sup>). However, their application at the electrode–tissue interface is hampered by poor electrical activity and biocompatibility. It should be noted that the electrical activity of the neural interface is represented by some electrochemical parameters (*e.g.*, impedance, CSC, CIL), so the high electron conductivity of the metal electrodes does not mean the electrode has good electrical activity. For example, the conductive polymer has lower conductivity than platinum, but higher electrical activity.<sup>74,75</sup> Therefore, a low conductivity material is not necessarily a low electroactive material, and it is undeniable that the augment in conductivity of the electrode may reinforce its electrical activity.<sup>55</sup> In the following part, we primarily focus on discussing how to improve the electrical activity and biocompatibility of metal electrodes.

**3.1.1.1. Nanometal electrode.** Owing to the pressing need for neural electrodes with a small size, large number of channels, and high electrode density, researchers have fabricated plentiful representative electrodes such as metal microwire electrodes (Fig. 5a).<sup>76</sup> Nevertheless, this electrode still has faced inevitable shortcomings: (i) the metal microwire may be



**Fig. 4** Schematic diagram of the charge transfer mechanism of neural electrode materials: (i) charge transfer by electrons, (ii) charge transfer by ions, and (iii) charge transfer by the synergistic effect of electrons and ions.



**Fig. 5** (a) Diagram of the microwire electrode. Reproduced with permission from ref. 76. Copyright 2014, Nature Publishing Group. (b) SEM image of the metal-based electrode. Reproduced with permission from ref. 77. Copyright 2004, IEEE. (c) Schematic illustration of the microprobe electrode array. Reproduced with permission from ref. 79. Copyright 2003, IOP publishing. (d) Schematic of the tripod PDMS bending structure and related strain distribution analysis using the finite element method, (e) the change of the electrode resistance under diverse deformations, (f) the image of the electrode array on the rat brain (i), and ECoG signals recorded in normal (ii) and epileptic (iii) rats. Reproduced with permission from ref. 80. Copyright 2015, Wiley-VCH.

partly bent during the implanting process, generating it in an inconclusive position after implantation; (ii) this type of electrode has rigid characteristics, after implantation, it will cause glial activation to trigger tissue inflammation; and (iii) the electrode exhibits low charge injection capability and relatively high impedance, which lead to a decrease of performance in electrophysiological recording and stimulation. Advances in materials and micromachining technology have accelerated the development of another type of metal-based electrode (Fig. 5b),<sup>77</sup> which is made of stainless steel/titanium. After the stainless steel electrode had been chemically etched to smooth the electrode surface, platinum was electro-deposited on its tip, which promotes the charge transfer at the electrode–tissue interface.<sup>78</sup> The dominating disadvantage of these electrodes is their incompatibility with integrated circuit fabricating technology. As researchers continue to optimize the performance of neural electrodes, nanoporous metal electrodes have been developed. Both *in vivo* and *in vitro* studies have found that these electrodes had lower impedance and higher conductivity than their smooth counterparts. For example, a porous silicon electrode implanted in the brain of a rat was not only closer to neurons than smooth silicon, but also activated less glial.<sup>81</sup> *In vitro* porous platinum electrodes have lower impedance than smooth platinum electrodes.<sup>82</sup> Moreover, Aifantis *et al.*<sup>83</sup> implemented a nanoporous tungsten electrode for signal recording in the cerebral cortex and hippocampus of healthy rats. Compared with smooth electrodes, it was revealed that nanoporous tungsten had superior consistency and less signal attenuation within 4 months of implantation.

Recently, to further improve the electrode performances, flexible substrates (*e.g.*, polydimethylsiloxane (PDMS), polyimide, polyurethane (PU), poly-*p*-xylene) have been introduced

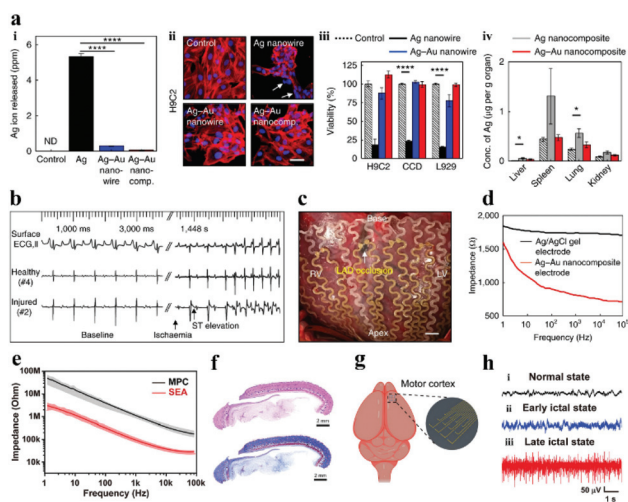
into the electrode, which represents the following advantages: (i) the electrode provides conformal adhesion to fit on the surface of brain/muscle to record stable signals and gain stimulation; (ii) flexible electrodes can record a diversity of neurological activities in a tenderly invasive manner with less injury; and (iii) the electrode can effectively decrease the impedance at the neural interface and reduce tissue injuries to further advance its electrical performance. As monitored in Fig. 5c, Fujita *et al.*<sup>79</sup> successfully designed and manufactured a three-dimensional flexible multi-channel microprobe electrode array. This electrode was successful in monitoring neural activity in disparate depth regions of the brain using poly-*p*-xylene as the substrate material. The flexible electrode can be well fitted on the brain surface and can shake synchronously with the weak movement of the brain, which supplies a powerful safeguard to avoid tissue harm during brain micro-movement. Afterwards, Qi *et al.* designed a high tensile gold nano-ribbon flexible electrode with a sinusoidal structure using a micro-tripod structured PDMS substrate (Fig. 5d).<sup>84</sup> The resistance of the electrode did not increase under large deformation (130% strain) (Fig. 5e), and the electrodes maintained stable conductivity during 10 000 cycles of the stretching/relaxing process, which is enough to avoid the distortion of the signal in the process of dynamic electrophysiological recording. The as-prepared electrodes were successful in recording and distinguishing the cortical electroencephalogram (ECoG) in normal rats and epileptic rats (Fig. 5f).

**3.1.1.2. Composite metal electrode.** The traditional metal electrodes have adverse effects to overcome for electrophysiological monitoring and tissue stimulation, such as high impedance and poor biocompatibility. Consequently, metal composite materials are proposed to improve the long-term performance of metal based neural electrodes. According to the design point of view, the generation of parasitic effects should be evaded, so the combination of similar materials is more preferred. Normally, metal materials and their derivatives (iridium oxide, gold, platinum) are selected as materials for the fabrication of composite metal electrodes.

As discussed previously, a large surface area of the electrode can reduce its impedance and enhance its charge injection capability. For metal electrodes, nano-structured materials with high porosity were used to improve the effective surface area of the neural electrode, thereby facilitating its electrochemical performance. Presently, gold (Au) is considered to be a hopeful material, which is normally employed for electrode coating of metals by electrochemical deposition, sputtering, thermal evaporation and so on. Kim *et al.* fabricated a multi-electrode array (60 channels) by coating gold nanoparticles (AuNPs) on the electrode using a bi-layer lift-off resist and sputtering deposition (SiO<sub>2</sub>) process.<sup>85</sup> They disclosed that the impedance of the modified electrode was reduced by four times compared with the original electrode. Subsequently, the researchers found that using nano-structured templates to prepare metal coatings is an excellent method to enlarge the effective surface area and reduce impedance.<sup>86,87</sup> Additionally, the good biocompatibility of AuNP coated electrodes has also

been proven.<sup>88,89</sup> Encouragingly, the electrode obtained by co-deposition of Ag–Au alloy has higher porosity, whose impedance was approximately 25 times lower than that of the pristine electrode, and the CSC and CIL of the electrode were equivalent to the values of an electrode made of titanium nitrous and carbon nanotubes, respectively.<sup>88–90</sup>

To overcome the rigid and brittle features of metal electrodes, Choi *et al.*<sup>91</sup> innovatively employed a gold shell layer to coat silver nanowires, and then combined it with a rubber elastomer (poly (styrene–butadiene–styrene)) to fabricate stretchable neural electrodes, and the as-prepared soft and highly stretchable Ag–Au core–sheath nanocomposite material achieved remarkable electrical conductivity ( $41\,850\text{ S cm}^{-1}$ ). The biocompatibility of the electrode in cell culture medium, human cells and mouse organs was much better than that of silver fibers (Fig. 6a), which is attributed to the deposition of a thick gold sheath on the surface of silver nanowires to prevent silver oxidation and ion leaching. The Ag–Au nanocomposite electrode not only minimizes the gap between the electrode and the skin, but also has a lower impedance (Fig. 6d), resulting in high-quality electrocardiogram (ECG) and electromyogram (EMG) signal monitoring (Fig. 6b and c). In this study, the gold core–shell structure was implemented to protect the silver fiber, which cleverly solved the incompatibility between the silver fiber and the biological environment. It provides a solution for similar problems in other related research territories.



**Fig. 6** (a) The biocompatibility of Ag–Au nanocomposite electrodes and silver fibers: the concentration of silver ions in the cell culture medium (i), the effects of different cell viability (ii and iii) and various organs in mice (iv), (b) electrode-recorded surface ECG and intracardiac electrograms of the ischemic heart in the healthy and injured areas, (c) photo of a cardiac mesh implanted on a pig heart, (d) the impedance of diverse electrodes at the electrode–skin interface. Reproduced with permission from ref. 91. Copyright 2018, Nature Publishing Group. (e) Impedance diagram of the electrode, (f) Hematoxylin & eosin and Masson staining results of the tissue surrounding the implanted SEA, (g) schematic of the recording of the SEA *in vivo*, (h) ECoG signals recorded by the SEA: (i) healthy, (ii) early and (iii) late epileptic rats. Reproduced with permission from ref. 92. Copyright 2021, Wiley-VCH.

Compared with the toxic silver nanowires,<sup>93,94</sup> liquid metal (LM) has good biocompatibility, excellent electrical conductivity and mechanical flexibility. Therefore, the liquid metal is constantly applied in the preparation of neural electrodes to conquer the challenges and deficiencies of traditional electrodes (*e.g.*, microwire electrodes, silicon-based electrodes, metal-based electrodes).<sup>95,96</sup> For example, Liu and his collaborators employed bismuth–indium–tin (Bi–In–Sn) liquid alloy materials to fabricate 3D microneedle electrodes *via* a micro-injection molding method.<sup>97</sup> After evaluating the electrical impedance and polarization voltage of the electrode, it is believed that the electrodes can meet the demands of practical applications. In addition, the electrode made from EGaIn (eutectic gallium indium) LM can not only be used *in vitro* to detect ECG physiological signals,<sup>98,99</sup> but also be implanted *in vivo* for neural activity monitoring. Recently, Dong *et al.* applied the screen printing technology to fabricate a stretchable electrode array (SEA) with EGaIn LM as the conductor and PDMS as the substrate, with a resolution of  $\sim 50\ \mu\text{m}$ .<sup>92</sup> The impedance of the electrode with Pt deposition is about  $250 \pm 40\ \text{k}\Omega$  at a frequency of 1 kHz in phosphate buffer solution (PBS), which is much lower than that of the liquid metal electrode ( $2.7\ \text{M}\Omega \pm 150\ \text{k}\Omega$ ) (Fig. 6e). What is more noteworthy is that one month after the SEA was implanted in the back of rats, the surrounding tissues had been almost unaffected (Fig. 6f), implying that the SEA has benign biocompatibility. The SEA was also implanted in the brain of rats and successfully monitored the real-time data of the rats in disparate states of epileptic seizures (Fig. 6g). The high-quality data (Fig. 6h) demonstrate that the electrodes formed conformal contact with the brain surface and the electrodes were competent for recording high-throughput ECoG signals. In parallel, the LM gallium indium tin (GaInSn) alloy has been considered as a promising material for restoring nerve function.<sup>95,100,101</sup> Currently, the thorny problems of LM in practical applications are as follows: (i) overcome the surface tension of LM, making it an ideal shape for neural electrodes; (ii) destroy the oxide layer of the LM particles inside the ink, so that the prepared electrode path can be connected; (iii) the embrittlement of the connection point between the LM and traditional conductive metal materials (*e.g.*, copper, gold, zinc) should be reduced, which can effectively improve the stability of the neural electrode; and (iv) the encapsulation of the LM electrode, which can increase the stability and solve the problem of leakage.

**3.1.2. Carbon-based flexible electrode.** The selection of materials has a great influence on the electrochemical performance, biocompatibility and stability of neural electrodes. Currently, carbon-based materials such as graphene and carbon nanotubes (CNTs) are considered outstanding candidates for the construction of flexible, miniaturized and multifunctional neural electrodes due to their excellent electrical, mechanical and chemical properties.

**3.1.2.1. Graphene electrode.** Graphene is a single-layer two-dimensional honeycomb lattice structure material formed by densely packed carbon atoms connected by  $\text{sp}^2$  hybridization. Recently, graphene has been seen as an executable material in

the territory of neural interfaces. Its preminent features include the following: (i) it is one of the highest-strength materials known, while still having good flexibility; (ii) it possesses attractive electronic conductivity (thin layer resistance:  $100 \Omega \text{ sq}^{-1}$ ) with a carrier mobility of  $15\,000 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  at room temperature (Fig. 7a);<sup>25,102</sup> (iii) it has a large specific surface area, which supplies a favorable environment for charge transfer and cell attachment, thereby enhancing the efficiency of stimulation and the sensitivity of recording; and (iv) it exhibits fascinating biocompatibility, which is ascribed to the functional groups modified on the graphene surface, making them more suitable for research in the field of biomaterials (Fig. 7b). Furthermore, graphene can be functionalized to produce graphene oxide, reduced graphene oxide and chemically modified derivatives, which provide more choices for its application as biological neural electrodes.<sup>103–106</sup> Hence, the preparation of flexible graphene microelectrodes is helpful to realize the tissue stimulation and neural activity

recording, owing to the high electrochemical performance of graphene.<sup>107</sup> For example, Yao *et al.*<sup>108</sup> prepared a flexible graphene microelectrode and treated it with the vapor plasma technology to form hydrophilic chemical bonds (*e.g.*, C–OH, C=O, OH–C=O) on the surface. This treatment promoted the decline of the interfacial resistance from 7216 to  $5424 \Omega \text{ mm}^{-2}$  and the reinforcement of the specific capacitance from  $0.7 \times 10^{-3}$  to  $1.4 \times 10^{-3} \text{ mF cm}^{-2}$ , respectively. The electrode has been successfully used for ECG monitoring in zebrafish.

The impedance of graphene fiber based neural electrodes could be dramatically reduced by coating Pt on the electrode surface.<sup>110</sup> After Pt coating, the impedance of this electrode was reduced by 5 times compared with the bare graphene electrode, and 300 times lower than that of the bare Pt electrode. Simultaneously, the charge injection capacity (CIC) of the electrode was increased from  $0.2 \text{ mC cm}^{-2}$  to  $10.34 \text{ mC cm}^{-2}$  (Fig. 7d). This is due to the good porous structure of graphene fibers prepared by the wet spinning method, which increases the surface area of the composite material. At the same time, the electrode successfully recorded the signal of a single neuron's activity at a depth of 1.5 mm below the motor cortex of rats (Fig. 7c), indicating the advantages of the electrode for *in vivo* neural recording. Accordingly, Garrett and his collaborators also gained graphene oxide fibers by wet spinning, and successfully made liquid crystal graphene oxide fibers by annealing.<sup>111</sup> Afterwards, the fiber end was disposed with a laser to render the surface roughness and porosity (Fig. 7e), and finally a neural electrode with high charge injection capability was fabricated. After implanting the electrode on the cat's visual cortex, the action potential of neural activity was successfully recorded (Fig. 7f). Otherwise, graphene can be used not only as a conductive layer, but also as a coating layer on neural electrodes to enhance the biocompatibility and electrochemical performance of the traditional neural electrodes.<sup>113,114</sup> The graphene coating layer can be pre-grown on the substrate by a spraying method<sup>115</sup> or a chemical vapor deposition (CVD) strategy<sup>116</sup> and then transferred to the target electrode. However, a challenge is that the stability between the coating graphene and the substrate is poor, leading to the degradation of the electrode performance. To address this challenge, Lyu's group employed laser cracking techniques to fabricate a porous graphene electrode array directly on a polyimide substrate.<sup>112</sup> The impedance of this electrode was about two orders of magnitude smaller than that of gold at the same size, which was due to the seamless contact between the substrate and the three-dimensional porous graphene. After the electrode was chemically doped with nitric acid, the impedance reduced to  $519 \Omega$  at 1 kHz (Fig. 7g) and the CIC increased from  $2 \text{ mC cm}^{-2}$  to  $3.1 \text{ mC cm}^{-2}$ . Furthermore, by placing the electrode array on the cortex of a rat, the evoked potential signals were successfully recorded (Fig. 7h). Simultaneously, the electrode was also successful in inducing the flexion of the ankle and knee joints by applying electrical stimulation.

**3.1.2.2. Carbon nanotube electrode.** CNTs can be regarded as one-dimensional nanomaterials formed by curling gra-



**Fig. 7** (a) Conductivity of graphene fibers at diverse temperatures. Reproduced with permission from ref. 25. Copyright 2020, Elsevier. (b) Diagram of graphene neural electrode arrays. Reproduced with permission from ref. 109. Copyright 2014, Nature Publishing Group. (c) 1543 single unit signals acquired by implanted electrodes, and the black wire in the center of the waveform expressing the average value of single unit signals, (d) plot of CIC, specific impedance, and geometric area compared with other neural electrodes. Reproduced with permission from ref. 110. Copyright 2019, Wiley-VCH. (e) Surface roughness and porosity generated after laser treatment of fiber ends, (f) neural activity recorded within 20 seconds of implantation into the visual cortex of cats. Reproduced with permission from ref. 111. Copyright 2015, Wiley-VCH. (g) Comparison chart of the electrochemical impedance of different materials, (h) spontaneous upward and downward state signals recorded by 16 electrode arrays. Reproduced with permission from ref. 112. Copyright 2016, Springer Nature.

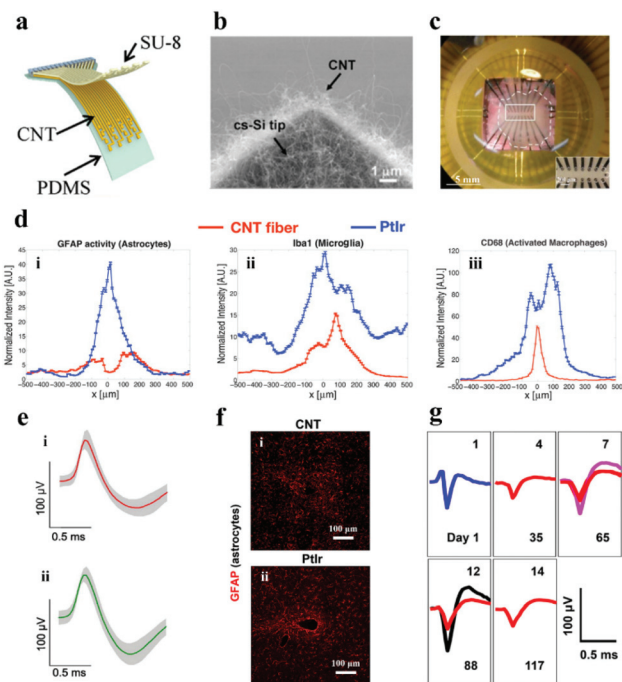


phene sheets with the following advantages for use in neural electrodes: (i) it has high conductivity, which facilitates the transfer of electrons from the interface to the electrode body; (ii) the material has a high surface area to volume ratio, which can promote the increase of the electrode charge injection ability and the decrease of the interface impedance;<sup>117,118</sup> and (iii) the functional groups contained on the surface of CNTs are easy to be modified by biomolecules, which enables the neural electrode to be adjusted in multiple directions according to the needs of practical applications.<sup>119,120</sup> Moreover, it also has perfect biocompatibility, flexibility, high mechanical strength, and captivating nerve cell adhesion.<sup>121–123</sup> Based on these ideal characteristics, CNTs have received more and more attraction in the orientation of neural electrodes (Fig. 8a).<sup>124–126</sup>

There is a trade-off between the impedance and the size of the neural electrodes. Reducing the size of the neural electrode always results in the increase of its impedance, however, the

electrode with a small size and low impedance is highly desired for neural activity recording with high resolutions. To balance this trade-off, Yew *et al.*<sup>127</sup> demonstrated a rewarding strategy for low impedance neural electrodes with a miniature size (3.5–65  $\mu\text{m}$  in diameter) by employing CNTs as the electrode material. Firstly, CNTs were grown on cone-shaped silicon by a CVD method (Fig. 8b), the cone-shaped silicon presents a larger surface area than the flat silicon, and the surface area was further enlarged by the deposited CNT. Therefore, the impedance ( $\sim 64.5 \Omega \text{ mm}^{-2}$ ) of the electrode was observably descended and the specific capacitance ( $\sim 2.5 \text{ mF cm}^{-2}$ ) was increased. The electrode has desired spatial resolution and splendid electrochemical performance for crayfish monitoring. The electrochemical properties of the electrode have been further improved with the consecutive investigation of the researchers,<sup>132–134</sup> but the flexibility has always been an emergency need for the practical application of neural electrodes.<sup>135</sup> Hanein and colleagues directly transferred the multi-walled CNT pattern grown by CVD to a flexible polymer substrate to fabricate a flexible full CNT electrode array.<sup>129</sup> After repeated folding and winding for 30 cycles, the resistivity of the electrode did not change too much. Moreover, the electrode was successful in extracellular neuron recording and the stimulation of the chicken retina (Fig. 8c). It manifests that the electrochemical performance and flexibility of neural electrodes with the assistance of CNTs can be signally strengthened. Moreover, CNTs also play an important role in improving the biocompatibility of the neural electrode. The CNT fiber electrode manufactured by Pasquali's team showed a lower impedance, and higher CSC ( $372 \pm 56 \text{ mC cm}^{-2}$ ) and CIL ( $6.52 \text{ mC cm}^{-2}$ ) compared with the PtIr based line-shaped electrode.<sup>130</sup> More importantly, the number of activated microglia and accumulated astrocytes at the CNT fiber electrode was twice and four times lower than that at the PtIr electrode after they were embedded in rats for 6 weeks (Fig. 8d). The CNT fiber electrode was successful in recording single neuron action potentials for at least 21 days after being implanted into the primary motor cortex of cats (Fig. 8e). It demonstrates that the CNT fiber electrode has better biocompatibility and stability. In addition, Duan *et al.* have made microelectrodes with small-diameter carbon nanotube fibers, which constituted a double electric layer at the electrode–solution interface to control the electrochemical process.<sup>131</sup> It delivered better CIL and CSC properties compared with the PtIr electrode. In addition, the CNT fiber microelectrodes have more stable and gainful microenvironmental ability in the neural interface when compared with the metal rigid electrode. Furthermore, the researchers implemented an innovative shuttle-assisted means to precisely position the CNT fiber electrode to the target brain region and recorded individual neurons for up to 4 months (Fig. 8g). Meaningfully, the electrode can be retracted controllably after insertion owing to its good tensile strength.

In addition to serving as a conductive layer, CNTs can also act as a modification material to improve the electrical activity of the neural electrode.<sup>134,136</sup> For instance, the impedance of the commercial tungsten neural electrode could be reduced



**Fig. 8** (a) Illustration of the flexible CNT electrode array. Reproduced with permission from ref. 128. Copyright 2018, American Chemical Society. (b) SEM image demonstrating the cone-shaped silicon tip of the CNT electrode. Reproduced with permission from ref. 127. Copyright 2010, Elsevier. (c) CNT electrode array mounted on the printed circuit board holder connects the electrodes to the exterior magnifier. Reproduced with permission from ref. 129. Copyright 2014, Springer. (d) Fluorescence intensity distribution when the lateral distance  $x$  of the electrode center line augments: (i) astrocytes, (ii) microglia and (iii) activated macrophages, (e) single neuron action potential data recorded on the (i) CNT fiber and (ii) NiCr-Au channel. Reproduced with permission from ref. 130. Copyright 2015, American Chemical Society. (f) Contrast of tissue inflammation between the CNT fiber and platinum iridium (PtIr) electrode, (g) average waveforms of individual neurons at detection and separation were recorded from day 1 to day 117. Reproduced with permission from ref. 131. Copyright 2019, American Chemical Society.

(from 940 k $\Omega$  to 38 k $\Omega$ ) after modifying CNTs on its surface, at the same time, the CSC of the electrode has enhanced approximately 40 times.<sup>137</sup> Moreover, the amplitude of the electrophysiological signal recorded by this electrode was larger and its noise was lower compared with that recorded by the unmodified electrode. Additionally, Vafaiee and his collaborators prepared a gold multi-electrode array coated with CNTs, and the impedance after CNT modification was reduced by 50%.<sup>121</sup> Not only that, the CNT coated electrode also was able to monitor the electrophysiological activity *in vitro*.<sup>138</sup>

### 3.2. Neural electrodes that transfer charge by ions

Signal transmission in human biological systems is dominated by the transfer of ions and small molecules, rather than electrons and holes in electronic devices.<sup>139,140</sup> When ions act as charge carriers in water-rich biological tissues, the tissues have much higher electrical conductivity than using electrons and holes as the charge carriers.<sup>141,142</sup> Consequently, ionically conductive bioelectronic devices have been paid more and more attention by scientific researchers.<sup>143–145</sup> Currently, the application of liquid-type ionic conductors (*e.g.*, electrolyte solutions and ionic liquids) is restrained due to the need for molds to maintain their shape.<sup>146</sup> In contrast, solid ionic conductors have attracted much more attention.<sup>147,148</sup> Recently, owing to the inherent remarkable biocompatibility, flexibility, biological activity, and compliance with nerve tissue, hydrogels have been widely used in the biological investigation including cell culture,<sup>149–151</sup> drug delivery,<sup>152–155</sup> tissue repair,<sup>156–158</sup> reconstruction, *etc.*<sup>159–161</sup> Simultaneously, hydrogels have also been employed in the manufacture of functional devices, such as sensors,<sup>163–165</sup> optics,<sup>165–167</sup> bioelectronics and so on. Importantly, as candidate materials for flexible bioelectronics, hydrogels have become a momentous part of neural electrodes.<sup>168–170</sup>

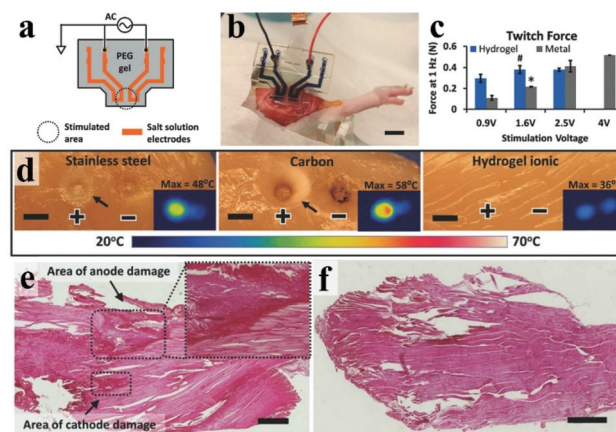
**3.2.1. Neural electrode made from hydrogels.** Hydrogels are not only a member of the soft substance family,<sup>171</sup> but also a solid type ion conductor. Hydrogels are composed of cross-linked polymer networks, in which the ions can freely move in the absorbed water, thus facilitating ionic conduction.<sup>15,19,21</sup> As highly biocompatible materials,<sup>168</sup> hydrogels have been used in artificial muscles,<sup>172</sup> ionic skin,<sup>173</sup> artificial axon devices and so on.<sup>174</sup> The functions of these devices were realized by ionic conduction. Ionic conduction mainly relies on a non-Faraday process without substances or charges passing through the interface, and it possesses the feature of transmitting high-frequency electrical signals over long distances.<sup>175</sup>

To our delight, three dominating hydrogels have been explored to be engaged in the electrode–tissue interface: (i) ion conductive hydrogels; (ii) ion conductive organohydrogels; and (iii) composite hydrogel materials. In the following part, the advantages of hydrogels in boosting the electrical activity of neural interfaces are discussed, which will furnish valuable reference strategies for the electrochemical properties and long-term stability of neural interfaces during the process of electrophysiological signal recording and neural stimulation. Here, we focus on hydrogels with ionic conductivity for neural

electrodes. The hydrogels whose conductivity is contributed by the synergistic effect of the electrons and ions will be discussed in 3.3.4.

**3.2.2. Neural electrode made from ion conductive hydrogels.** Ionic conductive hydrogels (ICHs) are a class of highly hydrophilic gels with a three-dimensional network structure.<sup>176</sup> Because free ions can move in a broadly hydrated network, the hydrogels can acquire high ionic conductivity (3.4–5.5 S m<sup>-1</sup>)<sup>177,178</sup> by permeating salts (including NaCl, LiCl, FeCl<sub>3</sub>, *etc.*), acids (including HCl, H<sub>3</sub>PO<sub>4</sub>, *etc.*) or ionic liquids.<sup>179–181</sup> The ion conductive feature of the hydrogels is similar to that of the biological tissues, so the hydrogel can profitably exchange material information with the surrounding tissues through ion diffusion,<sup>182</sup> and fundamentally evades the transformation between electron and ion signals and the derived problems.<sup>175,183</sup> Moreover, hydrogels are rich in a large amount of water, so even if side effects occur, they can be hired as a buffer medium to prevent adverse reactions from affecting the tissues. Notably, most of the ICH are transparent, adhesive and self-healing, these merits promote the applications of hydrogels in bioelectronics (*e.g.*, wearable devices, implantable electrodes and epidermal electrodes).<sup>184,185</sup>

However, ICHs prepared by incorporating salts have low stability and poor biocompatibility due to the diffusion of added ions, which are not suitable for bioelectronic devices. To solve this problem, Kaplan *et al.* developed a programmable ICH containing phase separation salt/polyethylene glycol (PEG) (Fig. 9a and b).<sup>162</sup> The advantage of this electrode is to ensure that the ionic aqueous solution will not be diffused



**Fig. 9** (a) Schematic diagram of an ionic stimulator electrode with ionic liquid encapsulated in a hydrogel, (b) photo of a salt/PEG hydrogel stimulated electrode in rat muscle, and scale bars are 1 cm, (c) the image of stimulating muscles to induce twitches using a 1 Hz pulse signal under the contrast conditions of standard gold electrodes, (d) using stainless steel, carbon and Alg-PAAm electrodes for current injection, the results reveal that Alg-PAAm electrodes can effectively lower local heating caused by current injection, and indirectly protect tissue damage, scale bars are 5 mm, (e) during the injection of high current, the gold electrode caused damage to the muscle tissue and (f) the Alg-PAAm ion electrode did not cause muscle injury, scale bars are 1 mm. Reproduced with permission from ref. 162. Copyright 2018, Wiley-VCH.

into the surrounding tissue media. Experiments show that the ICH ion electrode can produce complete twitch forces at a smaller stimulation voltage compared with the traditional metal electrode, indicating that the ICH ion electrode shows good performance for bioelectrical stimulation (Fig. 9c). As illustrated in Fig. 9d–f,† the ICH ion conductor is relatively stable in a physiological environment, indicating that the electrode can satisfy the long-term electrical stimulation of muscles in the body. Additionally, it has been reported<sup>186</sup> that implanted neural electrodes coated with polyethylene glycol dimethacrylate (PEG-DMA) hydrogels have preminent biocompatibility and can effectively decrease the injury of nerve cells around the electrodes. Nevertheless, in monitoring, unnecessary gaps are generated at the electrode–tissue interface due to muscle contraction or skin bending (surface electromyography abbreviation sEMG), which leads to serious noise/error in the detected signals, especially the signal from low-level muscle contraction. These gaps could be eliminated by introducing electrostatic interactions between the electrode and tissues. For example, Chen and his collaborators reported a compliant electrode based on alginate-polyacrylamide (Alg-PAAm) (Fig. 10a).<sup>51</sup> The electrode eliminated interfacial gaps by forming extensive hydrogen bonds on the skin surface (Fig. 10b and c). As a result, a low interfacial impedance (about 20 k $\Omega$  at 1 Hz *versus* approximately 500 k $\Omega$  of the commercial electrode) was achieved (Fig. 10d) and 2.1% of the maximum autonomic contraction of the muscle was successfully monitored.

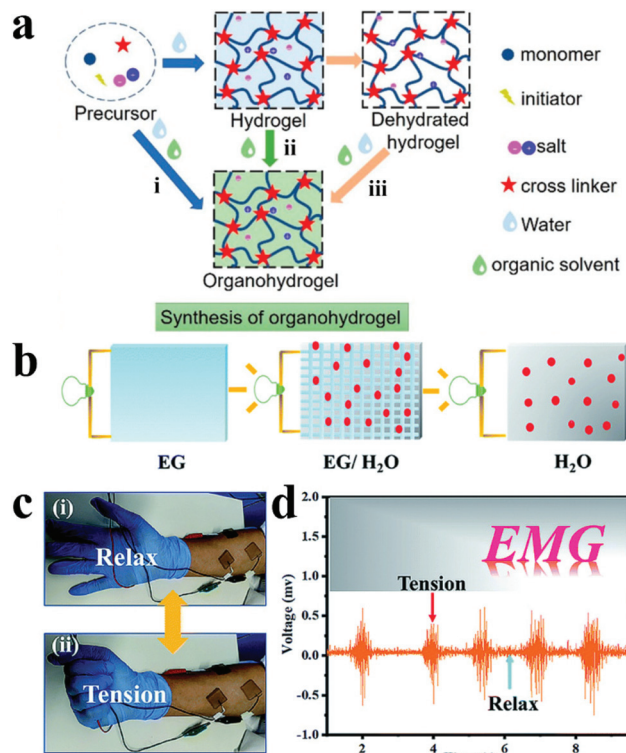


**Fig. 10** (a) Schematic of the electrode and skin used for EMG detection: describes the coupling process between ion flux (electrolyzed tissue medium) and current (electrode); SEM pictures display the interfacial clearance between pigskin and (b) commercial electrode or (c) Alg-PAAm electrode, respectively, (d) the impedance diagram of the electrode–skin system from 1–10<sup>4</sup> Hz. Reproduced with permission from ref. 51. Copyright 2020, Wiley-VCH.

Although ICHs have been widely used in bioelectronics, ICH based electrodes tend to lose flexibility, conductivity and even change their morphology due to the characteristics of rapid dehydration, which severely inhibits their practical applications.<sup>187</sup> At present, the solutions to this problem include adding dehydrating reagents,<sup>188,189</sup> mingling polymer-ionic liquid gels,<sup>190,191</sup> attaching sealing materials,<sup>192</sup> and blending deep eutectic solvents (DESS) Recently, Guo and his team synthesized a transparent anti-drying PAAC-DES gel by polymerising monomer acrylic acid (PAAc) in a DES through ultraviolet radiation.<sup>193</sup> The as-prepared gel showed high adhesion ( $\sim 100$  N m<sup>-1</sup>) and desirable electrical conductivity (1.26 mS cm<sup>-1</sup>). In addition, the PAAc-DES gel not merely exhibits captivating compatibility to the skin after 8 hours of wearing, but also has low impedance and records high-quality sEMG signals compared with the commercial electrodes (Ag–AgCl electrode).

**3.2.3. Neural electrode made from ion-conducting organo-hydrogels.** At present, dehydration is one of the biggest problems in practical applications of hydrogels. Recently, introducing appropriate organic solvents into hydrogels to obtain ion-conducting organohydrogels (ICOHs) paves an effective way to surmount this shortcoming. This is attributed to the fact that organic solvents can replace part of the water in the hydrogels to improve their dry immunity and maintain good ion conductivity.<sup>194</sup> To our delight, ICOHs retain the distinct advantages of hydrogels including biocompatibility,<sup>195,196</sup> shape/structure designability,<sup>197</sup> soft mechanical properties and so on.<sup>198,199</sup> The methods of introducing organic solvents can be divided into three types (Fig. 11a): (i) the desorbed hydrogel network is injected with organogel precursors, and then is subjected to *in situ* polymerization;<sup>200</sup> (ii) solvent replacement;<sup>201</sup> and (iii) gelation in a binary solvent.<sup>202</sup> Generally, solvent displacement is a convenient way to acquire ICOHs. However, the force between the polymer network in the hydrogel and the replacement solution is relatively weak, resulting in leakage of the substitution fluid, thereby injuring human tissues.<sup>204</sup> In contrast, ICOHs synthesized *in situ* by binary solvent has the advantage of remarkable electrical conductivity. For example, by using the binary solvent method, Liu *et al.*<sup>203</sup> explored an ICOH of polyvinyl alcohol-tannic acid@talc (PVA-TA@talc) (Fig. 11b), which had excellent mechanical properties, transparency and good electrical conductivity. In particular, due to the worthy strain sensitivity conferred by the ion transport channel, the electronic device can be used as a bioelectrode to collect important epidermal muscle signals (EMG). The electrode showed comparable performance compared with the commercial Ag/AgCl electrode in electrophysiological signal recording (Fig. 11c and d).

Low electrochemical impedance and high tissue adhesion are the requirements of ion-conducting materials for their application in bioelectrodes for neural interfacing. Hou *et al.* employed the hydrogen bonds between the chain of silk fibroin (SF) and poly (acrylamide-co-acrylic acid) covalent network to make SF uniformly dispersed to the hydrogel.<sup>205</sup> Sodium chloride was added to form salting-out and accelerate molecular chain entanglement of hydrogels to increase ionic



**Fig. 11** (a) Three synthetic methods of organohydrogels: (i) gelation in binary solvents, (ii) organic solvent replacement and (iii) organohydrogels polymerizing *in situ* on dehydrated hydrogels. Reproduced with permission from ref. 194. Copyright 2021, American Chemical Society. (b) Schematic illustration of ion conduction in multiple solvents, (c) monitor EMG signals under skin (i) relaxation and (ii) tension, (d) PVA-TA@talc organohydrogel detected EMG signal data. Reproduced with permission from ref. 203. Copyright 2019, the Royal Society of Chemistry.

conductivity (Fig. 12a). As a result, a low impedance ( $27.8 \Omega$ ) was achieved with the SF organohydrogel. In addition, the free catechol/pyrogallol groups of tannic acid permit the formation of hydrogen bonds at the interface between the SF organohydrogel and the substrate, which enhanced the adhesion between the SF organohydrogel and the epithelial tissue (Fig. 12b). Hence, this electrode is a robust neural interface tool for monitoring ECG, microvolt EEG (Fig. 12c) and EMG signals (Fig. 12g). Incredibly, the electrode can be engaged for the auditory brainstem response (ABR) (Fig. 12d–f) and can obtain the same ABR detection signal waveform as the invasive metal electrode (Fig. 12h), which is of great significance for the diagnosis and treatment of patients with hearing impairment. In the long term, ICOHs are still in their infancy as an implantable nerve electrode, and the practical application of this type of electrode needs to be improved.

### 3.3. Neural electrodes that transfer charge by the synergetic effect of electrons and ions

Although the materials using electrons (metals, semiconductors) or ions (hydrogels, organohydrogels) as the charge



**Fig. 12** (a) Schematic diagram of the chemical structure of the SF organohydrogel, and the illustrations show the ionic conductivity and self-adhesive properties of the electrodes, respectively, (b) the electrode conglutination principle and low electrode–epidermis interface impedance, (c) ECG, EMG and EEG signals detected by the SF organohydrogel electrode, and the following figure shows the equivalent circuit of the tissue/hydrogel electrode/device interface for bioelectronic detection, (d) schematic of the SF organohydrogel interface and invasive electrode in the way of acquiring signals, (e) schematic description of the generated and transmitted ABR signals, (f) schematic of the SF organohydrogel interface and metal electrode measuring ABR signals, (g) the EMG signal of the SF organohydrogel on the forearm under diversiform clamping forces, (h) ABR signals collected by the SF organohydrogel electrode and invasive metal electrode under 90 dB stimulation conditions. Reproduced with permission from ref. 205. Copyright 2022, Elsevier.

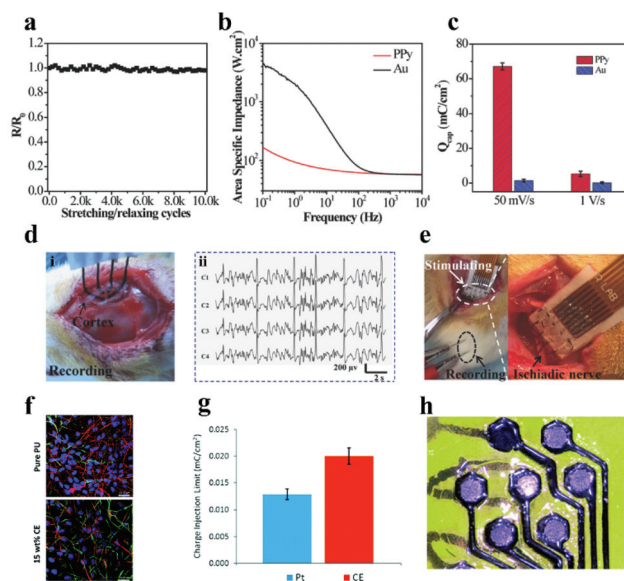
carrier have been explored for the construction of neural electrodes, the ideal candidate materials should satisfy the requirements for effective charge transfer between the external device (electrons as the charge carrier) and the tissue (ions as the charge carrier). Therefore, the materials with an electron–ion synergistic effect for charge transfer are more suitable for the actual development of nerve electrodes. Recently, conducting polymers have been widely used in neural electrodes because of their unique ion–electron synergistic charge transfer behavior. Moreover, composite materials have attracted much attention because they can integrate the advantages of each component and conquer the disadvantages of a single material.<sup>124,206,207</sup>

Conductive polymers (CPs) are frequently used in the research and development of neural electrodes due to the following characteristics: (i) CPs exhibit synergistic ionic and electronic conductivity,<sup>208</sup> and this characteristic can be used to reduce the electrochemical impedance of neural electrodes; (ii) their inherent adjustable mechanical properties render them more applicable for acting as bridge materials of the electrode–tissue interface; (iii) CPs have porous structures (nanoparticles or nanofibers), which can reinforce the specific surface area of the neural electrodes, thereby facilitating ion exchange between the recording site and the circumambient tissues; and (iv) the material is effortless to combine with polyelectrolytes (*e.g.*, polyacrylic acid, polymethacrylic acid, polystyrene sulfonic acid) and bioactive molecules,<sup>209</sup> which further improves the biocompatibility of the material.<sup>210</sup>

Thus, in this section, the neural electrodes based on intrinsic conductive polymers (*e.g.*, poly(3,4-ethylene-dioxythiophene) (PEDOT), polypyrrole (PPy), polyaniline (PANI) and so on) are discussed in detail (Fig. S1† for the conductive mechanism).<sup>67,68</sup> Then, we review the magnificent mission of integrating two or more electroactive conductive materials (examples include hydrogel composite and CP composite materials) to achieve the signal transmission at the interface between nerve electrodes and biological tissues. We hope that it could provide a reference solution for the manufacture of neural electrodes with excellent performance.

**3.3.1. Neural electrodes made from polypyrrole.** A large number of studies have demonstrated that polypyrrole and polythiophene derivatives, such as poly(3,4-ethylenedioxythiophene) doped poly(styrene sulfonate) (PEDOT:PSS) are diffusely implemented in neural electrodes. PPy is one of the CP materials most explored for the neural interface, which is attributed to its biocompatibility, high conductivity, effortless preparation and the properties of water solubility and low oxidation potential of its monomers.<sup>211,212</sup> For example, Anderson's team<sup>213</sup> developed a multi-electrode array based on a PPy/polyol-borate composite membrane with PPy as the conductor. Electrochemical impedance spectroscopy and cyclic voltammetry disclosed that the electrode had low impedance and admirable charge storage/injection ability. The electrode was successfully hired for EMG, ECG, ECoG, EEG signal detection and the stimulation of the denervated skeletal muscle. An ideal model of the CP based neural electrode is only using the CP without adding other conductive materials (*e.g.*, metals, CNT) to build all polymeric bioelectrode arrays for neural interfacing. To achieve this goal, the following issues should be considered: (i) high adhesion between the electrode and substrate material; (ii) conformal electrode–tissue interface; and (iii) excellent electrochemical performance. After addressing the above issues, Qi *et al.*<sup>214</sup> reported an all-polymeric neural electrode array by employing PPy as the conductive material. They prepared a nanowire modified PPy multielectrode array through an electropolymerization process. The nanowires could introduce a transition layer between the electrode and the substrate to enhance their adhesion (~1.9 MPa). There was an inconspicuous resistance change during 10 000 stretching/relaxation cycles, indicating that it had favourable stability (Fig. 13a). In addition, the thin compliant polymer electrode not merely supplied outstanding conformality at the electrode–tissue interface, but also had a lower impedance and higher CSC compared with gold electrodes under identical conditions (Fig. 13b and c). Under the premise of satisfying the mechanical and electrochemical properties, the electrode was successful in distinguishing the ECoG signal from normal and epileptic state rats (Fig. 13d and e), which is crucial for the clinical application of polymer bioelectronics.

**3.3.2 Neural electrodes made from poly(3,4-ethylenedioxythiophene) and its derivatives.** Nevertheless, PPy is prone to be irreversibly oxidized or easily interfered with under chemical conditions such as pH, which inhibits its evolution in



**Fig. 13** (a) Resistance changes of the microelectrode array after diverse stretching/relaxation cycles, (b) electrical impedance spectra of PPy membrane and Au membrane electrodes in phosphate buffer solution with pH = 6.8, (c) CSC of the PPy electrode and Au electrode, (d) monitoring photos (i) and ECG signal (ii) of the PPy electrode application test on epileptic rats, (e) electrode array-stimulated rat sciatic nerves (left) and magnified images (right). Reproduced with permission from ref. 215. Copyright 2017, Wiley-VCH. (f) Fluorescence image of the compatibility of pure PU and optimized 15 wt% CE electrode materials on human neural precursor cells after 7 days of culture, scale bar = 23.07  $\mu\text{m}$ , the red, green and blue in the picture represent neurons, astrocytes and nuclei, respectively, (g) CIL comparison data of the CE electrode array and Pt electrode, (h) CE organic flexible electrode array after laser ablation. Reproduced with permission from ref. 216. Copyright 2019, the Royal Society of Chemistry.

neural electrodes. Alternately, PEDOT is more favored than PPy due to the following factors: (i) PEDOT has a narrow band gap,<sup>217</sup> which promotes the shift of charges in the polymer chain; (ii) PEDOT has higher conductivity, which facilitates the preparation of neural electrodes with better electrical activity;<sup>55,67,218</sup> (iii) it has higher electrochemical stability, which is essential for obtaining a stable signal;<sup>219–221</sup> and (iv) PEDOT reveals better biocompatibility than PPy, which is an indispensable requirement for practical applications.<sup>222–225</sup> By adopting PEDOT:PSS as a conductive material, Blau *et al.*<sup>226</sup> prepared an extendable and non-cytotoxic all-polymer electrode array. This electrode showed good stability in capturing muscle's action potential *in vivo*. Although CPs have many unique advantages in bioelectronics, their fragile, brittle and stiff characteristics limit their application in neural electrodes. To solve this problem, Green *et al.*<sup>216</sup> fabricated a conductive elastomer based electrode by dispersing PEDOT:PSS into a soft elastomer (polyurethane, PU). The conductive elastomer (CE) electrode has better biocompatibility compared with pristine PU as demonstrated by 7 days of continuous culture of human neural precursor cells (ReNcell VM) (Fig. 13f). Moreover, the use of the laser micromachining technology can convert a CE

into a flexible organic electrode array (Fig. 13g), which has a higher CIL compared with the Pt electrode (Fig. 13h). Simultaneously, an all-polymer cuff electrode made of the CE was successfully developed,<sup>227</sup> which demonstrated good stability during manufacture, disinfection, cyclic tensile test, model wearing *in vitro* and so on. These studies demonstrate that a CE significantly improves the CIL performance of neural electrodes compared with commercial Pt ones. The increase in CIL promotes the preparation of miniaturized electrodes, which provides hope for the development of high-density microelectrode arrays.

**3.3.3. Neural electrodes made of materials composited with conductive polymers.** In addition to being hired as neural electrodes, CPs can also serve as a coating layer to modify the traditional metal electrodes. The performance of the decorated electrode can be improved in the following aspects: (i) the good electrochemical performance of CPs enables the modified metal electrode with a lower interface impedance; (ii) the soft mechanical properties of CPs can improve the interface contact between the electrode and tissue; and (iii) the good biocompatibility of CPs can decrease the immune rejection reaction of the tissues. For instance, Wang's research group synthesized a conductive PANI coating layer on the surface of the platinum electrode by a dipping method.<sup>228</sup> The PANI film was not damaged or fell off after sequential electrical stimulation for 30 days, indicating the good stability of the PANI coated platinum electrode. CPs can not only increase the stability of the composite electrode, but also improve the electrochemical performance of the electrode by using their intrinsic porosity. For example, the Abidian team discovered that the impedance of the gold electrode modified with PPy and PEDOT was descending by approximately 10-fold and 200-fold,<sup>229</sup> respectively. Furthermore, the impedance of iridium electrodes modified by PPy and PEDOT nanotubes decreased by about 24-fold and 187-fold, respectively.<sup>230</sup> Importantly, the charge storage capacity of the PEDOT modified electrode increased by approximately three orders of magnitude.

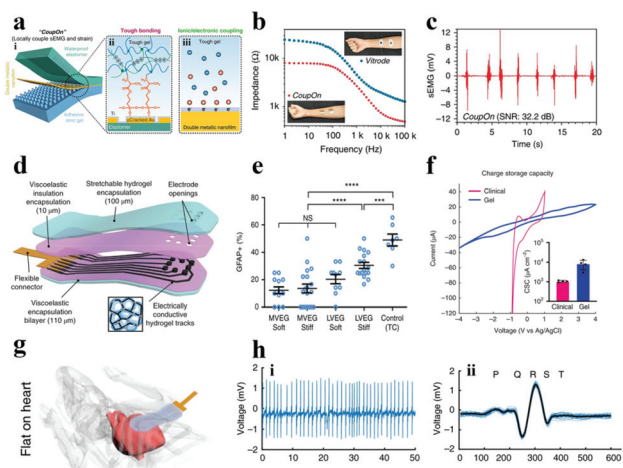
Additionally, carbon nanotubes (CNTs) have good mechanical strength and high conductivity (see section 3.1.2 for details). The combination of CPs with CNTs could enhance the conductivity of CPs and enlarge their surface areas. Hence, this CP/CNT composite material is a good candidate for modifying the metal bioelectrode. Thakor's group proposed a tunnel-like electrode coating layer made of PEDOT-CNTs for highly sensitive neural activity recording.<sup>233</sup> Compared with the bare gold electrode, the impedance of the modified gold electrode was reduced by 50-fold, and an increase (65-fold) of the CSC about the modified electrode was achieved. This may be due to the following reasons: (i) during the deposition process, CNTs act as a dopant, resulting in a strong interaction between PEDOT and CNTs (the adhesion of the PEDOT coating is increased by about 1.5 times), which promotes the rapid transmission of electrons within the composite material; and (ii) the formation of three-dimensional structure and the increase of the geometric area (approximately 1.8 times more

than non-nano tunnel structures). In parallel, a similar result was also achieved by Saunier *et al.*<sup>206</sup> They uncovered that the electrodeposition of a composite material (PEDOT and carbon nanofibers) on a gold microelectrode array could reduce the electrical impedance and enhance the charge storage capability of the electrode.

**3.3.4. Neural electrodes made of materials composited with hydrogels.** Hydrogels can generate ion current *via* the ions dissolved in water which have been absorbed in the polymer network of the hydrogels. It has commendable advantages in signal transmission at the electrode–tissue interface. To further promote the application of hydrogels in electronics, the key is to enable the hydrogels with the capability of transferring charge by electrons.<sup>148,234</sup> Fortunately, the porous structure of hydrogels furnishes sufficient space for diversification of conductive materials (*e.g.*, metal, carbon-based materials, CPs, *etc.*) to form conductive networks. Therefore, it is feasible to enhance the electrochemical properties (lower interface impedance and CIL) of hydrogels without sacrificing their ideal biological characteristics.<sup>235,236</sup>

CPs are widely used to construct composite materials with hydrogels for bioelectronics, owing to the following superiorities of CPs: (i) CPs have good flexibility and compatibility which benefits their hybridization with other polymer (hydrogel) systems; (ii) the unique organic and polymeric properties of CPs make them easy to be modified; and (iii) some CPs (*e.g.*, PEDOT) present a hydrogel form in humid environments, which are attributed to their hygroscopicity or swelling properties when exposed to water.<sup>21</sup> For example, a soft conductive hydrogel with high electrical activity was synthesized using PPy and alginate.<sup>237</sup> Its electrical conductivity increased by more than ten times compared with the original hydrogel without PPy. Moreover, the hydrogel compounded with PPy has good biocompatibility as it can be implanted in the mice for eight weeks without an obvious rejection reaction of the tissues. In addition, PEDOT was also successful upon compounding with hydrogels and enhancing its conductivity. For instance, a conductive hydrogel with adjustable electrical and physical (swelling, mechanical) properties was synthesized by employing PEDOT:PSS and bio-derived polymers (gelatin methacryloyl).<sup>238</sup> In particular, the impedance of hydrogels decreased from 449.0 k $\Omega$  to 281.2 and 261.0 k $\Omega$  after compounding 0.1% (w/v) and 0.3% (w/v) PEDOT:PSS at 1 Hz frequency, respectively.

Another way to enhance the conductivity of hydrogels is to compound electron conductive nanomaterials (*e.g.*, metal nanowires, LM, CNT, graphene, *etc.*) with hydrogels. The charge transfer in the resulting materials was contributed by both the ions and electrons,<sup>148,175</sup> which facilitates their application in bioelectrodes for neural interfacing. Inspired by the cytoadhesion structure, Cai *et al.* successfully prepared a locally coupled electromechanical interface based on a four-layer ionotronic hybridization (called CoupOn).<sup>231</sup> The CoupOn is a combination of bimetallic nanofilm (Au/Ti) deposited PDMS and hydrogel (alginate/polyacrylamide) (Fig. 14a). Fig. 14b shows that CoupOn has a lower impedance



**Fig. 14** (a) Schematic of the (i) coupled electromechanical interface of the four-layer ionization hybrid (*i.e.*, CoupOn), (ii) tough bonding, and (iii) ionic/electronic coupling; (b) impedance curves of the CoupOn and Vitrode electrode; and (c) sEMG of the CoupOn. Adapted with permission from ref. 231. Copyright 2020, Nature Publishing Group. (d) Illustration of the composition of the alginate hydrogel flexible electrode array material, (e) data graph recording primary cortical astrocytes on gels of disparate viscoelasticity (MVEG and LVEG) and stiffness (soft and stiff) in contrast to control tissue culture plastics (TC) after 120 h, where blue represents the nucleus, (f) CSC of viscoelastic electrode arrays and commercial grids, (g) diagram of the viscoelastic electrode array conforming to the mouse heart, (h) ECG of (i) mouse heart record and (ii) the average value of all beats superimposed (light blue displays a single cycle). Reproduced with permission from ref. 232. Copyright 2021, Nature Publishing Group.

than commercial electrodes (*i.e.*, Vitrode F150ML) in the entire frequency range, which furnishes a beneficial guarantee for sEMG signal monitoring (Fig. 14c). This strategy supplies a new scheme for the conversion of electron and ion signals at the electrode–tissue interface.

Besides, carbon-based materials have been one of the preferred materials for boosting the conductivity of ICH owing to the following advantages: (i) good conductivity, (ii) high surface area, and (iii) good stability in a wet environment. For example, Tringides *et al.*<sup>232</sup> successfully prepared electrode arrays with ultra-soft viscoelasticity (Fig. 14d), using ion-conductive sodium alginate and carbon nanomaterials (*i.e.*, CNT and graphene flakes) as a matrix and conductive additives, respectively. The softer and more viscoelastic gel (MVEG) not only has better biocompatibility than the harder and less viscoelastic gel (LVEG) (Fig. 14e), but also has a higher CSC compared to commercial grids (Fig. 14f). As monitored in Fig. 14g, the fully viscoelastic flexible electrode array can closely fit the surface of the heart and has appreciable ability in recording physiological signals (Fig. 14h). In short, under the premise of satisfying better electrochemical performance (Table S3†) and biocompatibility, the neural electrode, made from the materials transferring charge by ions and electrons, shows better signal transmission between the tissue and the external device.

## 4. Conclusion and outlook

Neural electrodes are key interfacial tools for electrophysiological communication between biological tissues and external electronic devices. Electrophysiological communication is a bidirectional process. In one way, the signals transfer from tissues, cross the electrode–tissue interfaces, deliver the electrode materials and finally are recorded by external devices. In reverse, the electrical signals transfer from the external devices to the tissues for stimulation. Thus, a good understanding of the transmission behavior of electrical signals in the tissues, the electrode–tissue interface and the electrode materials is helpful for building effective and stable neural electrode interfaces. In the tissues, the charge transfer was realized by ions, and the external electronic devices use electrons as the charge carriers to achieve high electrical conductivity. Therefore, neural electrodes with a synergistic charge transfer mechanism have more advantages in bidirectional information transfer. There are two ways to transmit electrical signals in the electrode–tissue interface. In one way, the signals cross the interface *via* electrical double layer capacitors. In the other way, the signal transfer is realized by a Faraday process. In brief, the signal transfer efficiency of the neural electrode depends on the impedance, charge storage capability, charge injection limitation and biocompatibility of the electrode.

On the one hand, the electrochemical performance of the neural electrode is mainly determined by interface impedance, CSC, CIL, *etc.* Too high impedance will cause resolution degradation and signal distortion. The methods of reducing impedance mainly include: (i) improving the conformal ability between the neural interface; (ii) improving the adhesion ability between the electrode and the tissue; and (iii) modifying the microstructure on the electrode surface. In addition, the low CSC and CIL will affect the charge transfer and lead to the weakening of the signal strength, which can be solved by enlarging the specific surface area of the electrode material.

On the other hand, biocompatibility is also a key factor affecting the quality of signal transmission, which can reduce the inflammatory reaction of the tissues and further hinder the formation of tissue scars, so as to achieve a close contact between the electrode and the tissue. The main methods to enhance the biocompatibility of the neural electrode include reducing the stiffness of the electrode materials, selecting biocompatible materials for the fabrication of neural electrodes, and coating the electrode with biocompatible layers.

Although neural electrodes have made rapid progress in body–machine communication in recent years, several challenges remain to be overcome before such electronic devices can be seamlessly integrated into practical applications: (i) the weak adhesion of the electrode–tissue interface limits the signal transmission. This may be due to the high hydrophobicity and low bioaffinity of the material. Therefore, the preparation of conformal electrodes is urgently needed; (ii) the poor compliance of the electrode–tissue interface is still a rigorous problem. It not only requires the electrode to have good stretchability/bending stiffness (reducing mechanical failure), but also to have a lower elastic modulus (decreasing foreign

body reaction), which ultimately avoids an increase in interfacial impedance and thus improves the electrode stimulation/recording efficiency; and (iii) the long-term stability of the electrode in the body still presents enormous challenges. This is not only due to the degradation of the electrode material in the body, but also due to mechanical delamination as a result of friction between the electrode and the tissue, resulting in a decrease in the electrical activity of the material.

In brief, neural electrodes are expected to achieve high efficiency, and precise and long term stable communication between the biological system and the external electronic systems in an electrical way. To realize this goal, we have to optimize the electrochemical performance and biocompatibility of neural electrodes by discovering new materials, designing novel micro/nanostructures and so on. Simultaneously, advanced microfabrication technology could promote the high throughput and accurate production of neural electrodes. It is believed that the development of neural electrodes could promote the evolution of neural-prosthesis, bioelectronic medicine, brain science, and so on.

## Conflicts of interest

The authors declare that they have no conflict of interest.

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## References

- 1 W. Wang, J. L. Collinger, M. A. Perez, E. C. Tyler-Kabara, L. G. Cohen, N. Birbaumer, S. W. Brose, A. B. Schwartz, M. L. Boninger and D. J. Weber, *Phys. Med. and Rehabil. Clin.*, 2010, **21**, 157–178.
- 2 E. H. Chang, *Bioelectron. Med.*, 2021, **7**, 1–3.
- 3 M. Zhang, Z. Tang, X. Liu and J. Van der Spiegel, *Nat. Electron.*, 2020, **3**, 191–200.
- 4 N. Xue, I. D. Martinez, J. Sun, Y. Cheng and C. Liu, *Biosens. Bioelectron.*, 2018, **112**, 114–119.
- 5 J. A. Frank, M. J. Antonini and P. Anikeeva, *Nat. Biotechnol.*, 2019, **37**, 1013–1023.
- 6 A. Curt, H. J. Van Hedel, D. Klaus, V. Dietz and E. S. S. group, *J. Neurotrauma*, 2008, **25**, 677–685.
- 7 C. M. Proctor, A. Slézia, A. Kaszas, A. Ghestem, I. Del Agua, A. M. Pappa, C. Bernard, A. Williamson and G. G. Malliaras, *Sci. Adv.*, 2018, **4**, eaau1291.
- 8 J. Lam, J. Lee, C. Y. Liu, A. M. Lozano and D. J. Lee, *Neuromodulation*, 2021, **24**, 171–186.
- 9 M. Heravi, K. Maghooli, F. N. Rahatabad and R. Rezaee, *Neurophysiology*, 2020, **52**, 375–387.
- 10 Y. Shi, R. Liu, L. He, H. Feng, Y. Li and Z. Li, *Smart Materials in Medicine*, 2020.
- 11 C. Sung, W. Jeon, K. S. Nam, Y. Kim, H. Butt and S. Park, *J. Mater. Chem. B*, 2020, **8**, 6624–6666.
- 12 G. A. Woods, N. J. Rommelfanger and G. Hong, *Matter*, 2020, **3**, 1087–1113.
- 13 M. Zhang, R. Guo, K. Chen, Y. Wang, J. Niu, Y. Guo, Y. Zhang, Z. Yin, K. Xia and B. Zhou, *Proc. Natl. Acad. Sci. U. S. A.*, 2020, **117**, 14667–14675.
- 14 W. M. Grill, S. E. Norman and R. V. Bellamkonda, *Annu. Rev. Biomed. Eng.*, 2009, **11**, 1–24.
- 15 M. Jia and M. Rolandi, *Adv. Healthcare Mater.*, 2020, **9**, 1901372.
- 16 L. Luan, J. T. Robinson, B. Aazhang, T. Chi, K. Yang, X. Li, H. Rathore, A. Singer, S. Yellapantula, Y. Fan, Z. Yu and C. Xie, *Neuron*, 2020, **108**, 302–321.
- 17 S. M. Wellman, J. R. Eles, K. A. Ludwig, J. P. Seymour, N. J. Michelson, W. E. McFadden, A. L. Vazquez and T. D. Y. Kozai, *Adv. Funct. Mater.*, 2018, **28**, 1701269.
- 18 W. M. Grill, S. E. Norman and R. V. Bellamkonda, *Annu. Rev. Biomed. Eng.*, 2009, **11**, 1–24.
- 19 M. Jia, J. Kim, T. Nguyen, T. Duong and M. Rolandi, *Biopolymers*, 2021, **112**, e23433.
- 20 J. Rivnay, H. Wang, L. Fenno, K. Deisseroth and G. G. Malliaras, *Sci. Adv.*, 2017, **3**, e1601649.
- 21 H. Yuk, B. Lu and X. Zhao, *Chem. Soc. Rev.*, 2019, **48**, 1642–1667.
- 22 K. Krukiewicz, J. Britton, D. Więclawska, M. Skorupa, J. Fernandez, J. R. Sarasua and M. J. Biggs, *Sci. Rep.*, 2021, **11**, 1–10.
- 23 F. E. Chan, H. M. Syu, T. Y. Wang, Z. T. Tang, C. N. Huang, J. F. Lee, T. Burnouf, S. H. Hu, P. C. Chen and W. C. Huang, *Adv. Mater. Interfaces*, 2021, 2100694.
- 24 R. A. Green, K. S. Lim, W. C. Henderson, R. T. Hassarati, P. J. Martens, N. H. Lovell and L. A. Poole-Warren, Living electrodes: tissue engineering the neural interface, in *2013 35th Annual International Conference of the IEEE Engineering in Medicine and Biology Society (EMBC)*, IEEE, 2013, pp. 6957–6960.
- 25 S. Liu, Y. Zhao, W. Hao, X. D. Zhang and D. Ming, *Biosens. Bioelectron.*, 2020, **170**, 112645.
- 26 A. S. Pranti, A. Schander, A. Bödecker and W. Lang, *Sens. Actuators, B*, 2018, **275**, 382–393.
- 27 R. H. Dong, L. L. Wang, C. Hang, Z. Chen, X. Y. Liu, L. N. Zhong, J. Qi, Y. Q. Huang, S. Q. Liu, L. P. Wang, Y. Lu and X. Y. Jiang, *Small*, 2021, **17**, 2006612.
- 28 L. Guo, M. M. Ma, N. Zhang, R. Langer and D. G. Anderson, *Adv. Mater.*, 2014, **26**, 1427–1433.
- 29 K. Tybrandt, D. Khodagholy, B. Dielacher, F. Stauffer, A. F. Renz, G. Buzsaki and J. Voros, *Adv. Mater.*, 2018, **30**, 1706520.
- 30 H. Wark, R. Sharma, K. Mathews, E. Fernandez, J. Yoo, B. Christensen, P. Tresco, L. Rieth, F. Solzbacher and R. Normann, *J. Neural Eng.*, 2013, **10**, 045003.



- 31 Y. M. Chen, T. W. Chung, P. W. Wu and P. C. Chen, *J. Alloys Compd.*, 2017, **692**, 339–345.
- 32 D. Qi, Y. Liu, Z. Liu, L. Zhang and X. Chen, *Adv. Mater.*, 2017, **29**, 1602802.
- 33 C. A. R. Chapman, N. Goshi and E. Seker, *Adv. Funct. Mater.*, 2018, **28**, 1703523.
- 34 Q. Yang, T. Wei, R. T. Yin, M. Wu, Y. Xu, J. Koo, Y. S. Choi, Z. Xie, S. W. Chen and I. Kandela, *Nat. Mater.*, 2021, **20**, 1559–1570.
- 35 M. Zhang, Z. Tang, X. Liu and J. Van der Spiegel, *Nat. Electron.*, 2020, **3**, 191–200.
- 36 A. R. Harris and G. G. Wallace, *Curr. Opin. Electrochem.*, 2019, **16**, 143–148.
- 37 D. T. Simon, E. O. Gabriellsson, K. Tybrandt and M. Berggren, *Chem. Rev.*, 2016, **116**, 13009–13041.
- 38 P. Fattahi, G. Yang, G. Kim and M. R. Abidian, *Adv. Mater.*, 2014, **26**, 1846–1885.
- 39 R. Biran, D. C. Martin and P. A. Tresco, *Exp. Neurol.*, 2005, **195**, 115–126.
- 40 J. W. Jeong, W. H. Yeo, A. Akhtar, J. J. S. Norton, Y. J. Kwack, S. Li, S. Y. Jung, Y. Su, W. Lee, J. Xia, H. Cheng, Y. Huang, W. S. Choi, T. Bretl and J. A. Rogers, *Adv. Mater.*, 2013, **25**, 6839–6846.
- 41 Y. Xie, N. Martini, C. Hassler, R. D. Kirch, T. Stieglitz, A. Seifert and U. G. Hofmann, *Front. Neuroeng.*, 2014, **7**, 34.
- 42 J. W. Salatino, K. A. Ludwig, T. D. Kozai and E. K. Purcell, *Nat. Biomed. Eng.*, 2017, **1**, 862–877.
- 43 E. R. Riva and S. Micera, *Bioelectron. Med.*, 2021, **7**, 1–10.
- 44 K. Woepfel, Q. Yang and X. T. Cui, *Curr. Opin. Biomed. Eng.*, 2017, **4**, 21–31.
- 45 S. M. Wellman, J. R. Eles, K. A. Ludwig, J. P. Seymour, N. J. Michelson, W. E. McFadden, A. L. Vazquez and T. D. Kozai, *Adv. Funct. Mater.*, 2018, **28**, 1701269.
- 46 A. Sridharan, S. D. Rajan and J. Muthuswamy, *J. Neural Eng.*, 2013, **10**, 066001.
- 47 C. Boehler, D. M. Vieira, U. Egert and M. Asplund, *ACS Appl. Mater. Interfaces*, 2020, **12**, 14855–14865.
- 48 M. Skorupa, D. Wicelawska, D. Czerwinska-Glowka, M. Skonieczna and K. Krukiewicz, *Polymers*, 2021, **13**, 1948.
- 49 Y. H. Kim, N. S. Baek, Y. H. Han, M. A. Chung and S. D. Jung, *J. Neurosci. Methods*, 2011, **202**, 38–44.
- 50 W. He and R. V. Bellamkonda, *Biomaterials*, 2005, **26**, 2983–2990.
- 51 L. Pan, P. Cai, L. Mei, Y. Cheng, Y. Zeng, M. Wang, T. Wang, Y. Jiang, B. Ji, D. Li and X. Chen, *Adv. Mater.*, 2020, **32**, 2003723.
- 52 J. H. Kim, S. R. Kim, H. J. Kil, Y. C. Kim and J. W. Park, *Nano Lett.*, 2018, **18**, 4531–4540.
- 53 C. A. Chapman, H. Chen, M. Stamou, J. Biener, M. M. Biener, P. J. Lein and E. Seker, *ACS Appl. Mater. Interfaces*, 2015, **7**, 7093–7100.
- 54 E. Kolaya and B. L. Firestein, *Biotechnol. Prog.*, 2021, e3179.
- 55 N. Chen, L. Tian, A. C. Patil, S. Peng, I. H. Yang, N. V. Thakor and S. Ramakrishna, *Nano Today*, 2017, **14**, 59–83.
- 56 E. Mtui, G. Gruener and M. T. FitzGerald, *Clinical Neuroanatomy and Neuroscience E-Book*, Elsevier Health Sciences, 2011.
- 57 P. S. Kaeser, L. Deng, Y. Wang, I. Dulubova, X. Liu, J. Rizo and T. C. Südhof, *Cell*, 2011, **144**, 282–295.
- 58 E. R. Kandel, J. H. Schwartz, T. M. Jessell, S. A. Siegelbaum and A. J. Hudspeth, *Principles of Neural Science*, 5th edn, 2012.
- 59 S. F. Cogan, *Annu. Rev. Biomed. Eng.*, 2008, **10**, 275–309.
- 60 D. R. Merrill, in *Implantable Neural Prostheses 2: Techniques and Engineering Approaches*, ed. D. Zhou and E. Greenbaum, Springer New York, New York, NY, 2010.
- 61 D. R. Merrill, M. Bikson and J. G. Jefferys, *J. Neurosci. Methods*, 2005, **141**, 171–198.
- 62 R. Dong, L. Wang, C. Hang, Z. Chen, X. Liu, L. Zhong, J. Qi, Y. Huang, S. Liu and L. Wang, *Small*, 2021, **17**, 2006612.
- 63 Y. H. Kim, H. Koo, M. S. Kim and S. D. Jung, *Sens. Actuators, B*, 2018, **273**, 718–725.
- 64 R. Thakur, A. R. Nair, A. Jin and G. Y. Fridman, Fabrication of a Self-Curling Cuff with a Soft, Ionically Conducting Neural Interface, in *2019 41st Annual International Conference of the IEEE Engineering in Medicine and Biology Society (EMBC)*, IEEE, 2019, pp. 3750–3753.
- 65 R. Green, *Nat. Biomed. Eng.*, 2019, **3**, 9–10.
- 66 C. Wu, A. Liu, S. Chen, X. Zhang, L. Chen, Y. Zhu, Z. Xiao, J. Sun, H. Luo and H. Fan, *ACS Appl. Mater. Interfaces*, 2019, **11**, 22152–22163.
- 67 N. K. Guimard, N. Gomez and C. E. Schmidt, *Prog. Polym. Sci.*, 2007, **32**, 876–921.
- 68 E. Zeglio, A. L. Rutz, T. E. Winkler, G. G. Malliaras and A. Herland, *Adv. Mater.*, 2019, **31**, 1806712.
- 69 R. Liu, S. Zhao and J. Liu, *ACS Appl. Electron. Mater.*, 2020, **3**, 101–118.
- 70 G. Hong, T. M. Fu, M. Qiao, R. D. Viveros, X. Yang, T. Zhou, J. M. Lee, H. G. Park, J. R. Sanes and C. M. Lieber, *Science*, 2018, **360**, 1447–1451.
- 71 T. Rose and L. Robblee, *IEEE Trans. Biomed. Eng.*, 1990, **37**, 1118–1120.
- 72 D. Borkholder, J. Bao, N. Maluf, E. Perl and G. Kovacs, *J. Neurosci. Methods*, 1997, **77**, 61–66.
- 73 S. F. Cogan, P. R. Troyk, J. Ehrlich and T. D. Plante, *IEEE Trans. Biomed. Eng.*, 2005, **52**, 1612–1614.
- 74 R. Balint, N. J. Cassidy and S. H. Cartmell, *Acta Biomater.*, 2014, **10**, 2341–2353.
- 75 S. Baek, R. A. Green and L. A. Poole Warren, *J. Biomed. Mater. Res., Part A*, 2014, **102**, 2743–2754.
- 76 D. A. Schwarz, M. A. Lebedev, T. L. Hanson, D. F. Dimitrov, G. Lehew, J. Meloy, S. Rajangam, V. Subramanian, P. J. Ifft, Z. Li, A. Ramakrishnan, A. Tate, K. Z. Zhuang and M. A. L. Nicolelis, *Nat. Methods*, 2014, **11**, 670–676.
- 77 T. A. Fofonoff, S. M. Martel, N. G. Hatsopoulos, J. P. Donoghue and I. W. Hunter, *IEEE Trans. Biomed. Eng.*, 2004, **51**, 890–895.
- 78 B. Ghane-Motlagh and M. Sawan, A review of microelectrode array technologies: design and implementation challenges, in *2013 2nd International Conference on Advances in Biomedical Engineering*, IEEE, 2013, pp. 38–41.

- 79 S. Takeuchi, T. Suzuki, K. Mabuchi and H. Fujita, *J. Micromech. Microeng.*, 2003, **14**, 104–107.
- 80 D. Qi, Z. Liu, M. Yu, Y. Liu, Y. Tang, J. Lv, Y. Li, J. Wei, B. Liedberg, Z. Yu and X. Chen, *Adv. Mater.*, 2015, **27**, 3145–3151.
- 81 K. Moxon, S. Hallman, A. Aslani, N. Kalkhoran and P. Lelkes, *J. Biomater. Sci., Polym. Ed.*, 2007, **18**, 1263–1281.
- 82 M. Li, Q. Zhou and Y. Y. Duan, *Sens. Actuators, B*, 2015, **221**, 179–186.
- 83 F. Shuang, H. Deng, A. B. Shafique, S. Marsh, D. Treiman, K. Tsakalis and K. E. Aifantis, *Mater. Lett.*, 2020, **260**, 126885.
- 84 D. Qi, Z. Liu, M. Yu, Y. Liu, Y. Tang, J. Lv, Y. Li, J. Wei, B. Liedberg, Z. Yu and X. Chen, *Adv. Mater.*, 2015, **27**, 3145–3151.
- 85 Y. H. Kim, G. H. Kim, N. S. Baek, Y. H. Han, A. Y. Kim, M. A. Chung and S. D. Jung, *J. Micromech. Microeng.*, 2013, **23**, 097001.
- 86 H. Yoon, P. Hankins, S. Oh, R. E. Harbaugh and V. K. Varadan, *J. Nanotechnol. Eng. Med.*, 2010, **1**, 021006.
- 87 V. Urbanová, Y. Li, K. Vytrás, B. Yvert and A. Kuhn, *J. Electroanal. Chem.*, 2011, **656**, 91–95.
- 88 C. A. R. Chapman, H. Chen, M. Stamou, J. Biener, M. M. Biener, P. J. Lein and E. Seker, *ACS Appl. Mater. Interfaces*, 2015, **7**, 7093–7100.
- 89 R. Zhang, D. Yao, P. A. Valdés-Sosa, F. Li, P. Li, T. Zhang, T. Ma, Y. Li and P. Xu, *J. Neural Eng.*, 2015, **12**, 066024.
- 90 E. Seker, Y. Berdichevsky, M. R. Begley, M. L. Reed, K. J. Staley and M. L. Yarmush, *Nanotechnology*, 2010, **21**, 125504.
- 91 S. Choi, S. I. Han, D. Jung, H. J. Hwang, C. Lim, S. Bae, O. K. Park, C. M. Tschabrunn, M. Lee, S. Y. Bae, J. W. Yu, J. H. Ryu, S. W. Lee, K. Park, P. M. Kang, W. B. Lee, R. Nezafat, T. Hyeon and D. H. Kim, *Nat. Nanotechnol.*, 2018, **13**, 1048–1056.
- 92 R. Dong, L. Wang, C. Hang, Z. Chen, X. Liu, L. Zhong, J. Qi, Y. Huang, S. Liu, L. Wang, Y. Lu and X. Jiang, *Small*, 2021, **17**, 2006612.
- 93 L. D. Scanlan, R. B. Reed, A. V. Loguinov, P. Antczak, A. Tagmount, S. Aloni, D. T. Nowinski, P. Luong, C. Tran, N. Karunaratne, D. Pham, X. X. Lin, F. Falciani, C. P. Higgins, J. F. Ranville, C. D. Vulpe and B. Gilbert, *ACS Nano*, 2013, **7**, 10681–10694.
- 94 I. G. Theodorou, K. H. Müller, S. Chen, A. E. Goode, V. Yufit, M. P. Ryan and A. E. Porter, *ACS Biomater. Sci. Eng.*, 2017, **3**, 2336–2347.
- 95 Q. Wang, Y. Yu and J. Liu, *Adv. Eng. Mater.*, 2018, **20**, 1700781.
- 96 C. Jin, J. Zhang, X. Li, X. Yang, J. Li and J. Liu, *Sci. Rep.*, 2013, **3**, 1–7.
- 97 R. Liu, X. Yang, C. Jin, J. Fu, W. Chen and J. Liu, *Appl. Phys. Lett.*, 2013, **103**, 193701.
- 98 Y. Yu, J. Zhang and J. Liu, *PLoS One*, 2013, **8**, e58771.
- 99 P. H. Kuo, T. H. Tzeng, Y. C. Huang, Y. H. Chen, Y. C. Chang, Y. L. Ho, J. T. Wu, H. H. Lee, P. J. Lai and K. Y. Liu, *PLoS One*, 2014, **9**, e104543.
- 100 J. Zhang, L. Sheng, C. Jin and J. Liu, *arXiv preprint arXiv:1404.5931*, 2014.
- 101 F. Liu, Y. Yu, L. Yi and J. Liu, *Sci. Bull.*, 2016, **61**, 939–947.
- 102 S. De and J. N. Coleman, *ACS Nano*, 2010, **4**, 2713–2720.
- 103 H. C. Tian, J. Q. Liu, D. X. Wei, X. Y. Kang, C. Zhang, J. C. Du, B. Yang, X. Chen, H. Y. Zhu and Y. N. NuLi, *Biomaterials*, 2014, **35**, 2120–2129.
- 104 Y. Lu, X. Liu, R. Hattori, C. Ren, X. Zhang, T. Komiyama and D. Kuzum, *Adv. Funct. Mater.*, 2018, **28**, 1800002.
- 105 Y. Lu, X. Liu and D. Kuzum, *Curr. Opin. Biomed. Eng.*, 2018, **6**, 138–147.
- 106 S. K. Lee, H. Kim and B. S. Shim, *Carbon Lett.*, 2013, **14**, 63–75.
- 107 S. Zhao, G. Li, C. Tong, W. Chen, P. Wang, J. Dai, X. Fu, Z. Xu, X. Liu and L. Lu, *Nat. Commun.*, 2020, **11**, 1–12.
- 108 C. H. Chen, C. T. Lin, W. L. Hsu, Y. C. Chang, S. R. Yeh, L. J. Li and D. J. Yao, *Nanomedicine*, 2013, **9**, 600–604.
- 109 D. Kuzum, H. Takano, E. Shim, J. C. Reed, H. Juul, A. G. Richardson, J. de Vries, H. Bink, M. A. Dichter, T. H. Lucas, D. A. Coulter, E. Cubukcu and B. Litt, *Nat. Commun.*, 2014, **5**, 5259.
- 110 K. Wang, C. L. Frewin, D. Esrafilzadeh, C. Yu, C. Wang, J. J. Pancrazio, M. Romero-Ortega, R. Jalili and G. Wallace, *Adv. Mater.*, 2019, **31**, 1805867.
- 111 N. V. Apollo, M. I. Maturana, W. Tong, D. A. X. Nayagam, M. N. Shivdasani, J. Foroughi, G. G. Wallace, S. Prawer, M. R. Ibbotson and D. J. Garrett, *Adv. Funct. Mater.*, 2015, **25**, 3551–3559.
- 112 Y. Lu, H. Lyu, A. G. Richardson, T. H. Lucas and D. Kuzum, *Sci. Rep.*, 2016, **6**, 1–9.
- 113 A. Bourrier, P. Shkorbatova, M. Bonizzato, E. Rey, Q. Barraud, G. Courtine, R. Othmen, V. Reita, V. Bouchiat and C. Delacour, *Adv. Healthcare Mater.*, 2019, **8**, 1801331.
- 114 C. Chiu, X. He and H. Liang, *Electrochim. Acta*, 2013, **94**, 42–48.
- 115 L. J. Wang, L. Li, J. Yu, Y. Wu, H. He, X. Ouyang, X. Zhao, Y. C. Yen and L. J. Lee, *Carbon*, 2014, **79**, 294–301.
- 116 B. Koerbitzer, P. Krauss, C. Nick, S. Yadav, J. J. Schneider and C. Thielemann, *2D Mater.*, 2016, **3**, 024004.
- 117 M. Shein, A. Greenbaum, T. Gabay, R. Sorkin, M. David-Pur, E. Ben-Jacob and Y. Hanein, *Biomed. Microdevices*, 2009, **11**, 495–501.
- 118 M. Meyyappan, *Carbon nanotubes: science and applications*, CRC press, 2004.
- 119 M. Bottini, N. Rosato and N. Bottini, *Biomacromolecules*, 2011, **12**, 3381–3393.
- 120 C. R. Butson and C. C. McIntyre, *Clin. Neurophysiol.*, 2005, **116**, 2490–2500.
- 121 M. Vafaiee, R. Mohammadpour, M. Vossoughi, E. Asadian, M. Janahmadi and P. Sasanpour, *Front. Bioeng. Biotechnol.*, 2021, **8**, 1465.
- 122 T. B. Nguyen-Vu, H. Chen, A. M. Cassell, R. J. Andrews, M. Meyyappan and J. Li, *IEEE Trans. Biomed. Eng.*, 2007, **54**, 1121–1128.

- 123 A. A. John, A. P. Subramanian, M. V. Vellayappan, A. Balaji, H. Mohandas and S. K. Jaganathan, *Int. J. Nanomed.*, 2015, **10**, 4267.
- 124 N. Chen, B. Luo, A. C. Patil, J. Wang, G. G. L. Gammad, Z. Yi, X. Liu, S. C. Yen, S. Ramakrishna and N. V. Thakor, *ACS Nano*, 2020, **14**, 8059–8073.
- 125 K. Foremny, W. S. Konerding, A. Behrens, P. Baumhoff, U. P. Froriep, A. Kral and T. Doll, *Nanomaterials*, 2021, **11**, 1029.
- 126 C. M. Lin, Y. T. Lee, S. R. Yeh and W. Fang, *Biosens. Bioelectron.*, 2009, **24**, 2791–2797.
- 127 H. C. Su, C. M. Lin, S. J. Yen, Y. C. Chen, C. H. Chen, S. R. Yeh, W. Fang, H. Chen, D. J. Yao, Y. C. Chang and T. R. Yew, *Biosens. Bioelectron.*, 2010, **26**, 220–227.
- 128 J. Zhang, X. Liu, W. Xu, W. Luo, M. Li, F. Chu, L. Xu, A. Cao, J. Guan, S. Tang and X. Duan, *Nano Lett.*, 2018, **18**, 2903–2911.
- 129 M. David-Pur, L. Bareket-Keren, G. Beit-Yaakov, D. Raz-Prag and Y. Hanein, *Biomed. Microdevices*, 2014, **16**, 43–53.
- 130 F. Vitale, S. R. Summerson, B. Aazhang, C. Kemere and M. Pasquali, *ACS Nano*, 2015, **9**, 4465–4474.
- 131 L. Lu, X. Fu, Y. Liew, Y. Zhang, S. Zhao, Z. Xu, J. Zhao, D. Li, Q. Li, G. B. Stanley and X. Duan, *Nano Lett.*, 2019, **19**, 1577–1586.
- 132 K. Wang, R. Y. Tang, X. B. Zhao, J. J. Li, Y. R. Lang, X. X. Jiang, H. J. Sun, Q. X. Lin and C. Y. Wang, *Nanoscale*, 2015, **7**, 18677–18685.
- 133 C. M. Voge and J. P. Stegemann, *J. Neural Eng.*, 2011, **8**, 011001.
- 134 A. Ansaldo, E. Castagnola, E. Maggolini, L. Fadiga and D. Ricci, *ACS Nano*, 2011, **5**, 2206–2214.
- 135 H. Li, J. Wang and Y. Fang, *Nanoscale Adv.*, 2020, **2**, 3095–3102.
- 136 A. Shoval, C. Adams, M. David-Pur, M. Shein, Y. Hanein and E. Sernagor, *Front. Neuroeng.*, 2009, **2**, 4.
- 137 E. W. Keefer, B. R. Botterman, M. I. Romero, A. F. Rossi and G. W. Gross, *Nat. Nanotechnol.*, 2008, **3**, 434–439.
- 138 G. Piret, C. Hébert, J. P. Mazellier, L. Rousseau, E. Scorsone, M. Cottance, G. Lissorgues, M. O. Heuschkel, S. Picaud, P. Bergonzo and B. Yvert, *Biomaterials*, 2015, **53**, 173–183.
- 139 M. Levin, G. Pezzulo and J. M. Finkelstein, *Annu. Rev. Biomed. Eng.*, 2017, **19**, 353–387.
- 140 Y. J. Jo, J. Ok, S. Y. Kim and T. I. Kim, *Adv. Mater. Technol.*, 2021, 2001273.
- 141 C. Cea, G. D. Spyropoulos, P. Jastrzebska-Perfect, J. J. Ferrero, J. N. Gelinas and D. Khodagholy, *Nat. Mater.*, 2020, **19**, 679–686.
- 142 K. Feron, R. Lim, C. Sherwood, A. Keynes, A. Brichta and P. C. Dastoor, *Int. J. Mol. Sci.*, 2018, **19**, 2382.
- 143 M. Jia, H. Dechiruji, J. Selberg, P. Pansodtee, J. Mathews, C. Wu, M. Levin, M. Teodorescu and M. Rolandi, *APL Mater.*, 2020, **8**, 091106.
- 144 T. Arbring Sjöström, M. Berggren, E. O. Gabrielsson, P. Janson, D. J. Poxson, M. Seitanidou and D. T. Simon, *Adv. Mater. Technol.*, 2018, **3**, 1700360.
- 145 H. R. Lee, C. C. Kim and J. Y. Sun, *Adv. Mater.*, 2018, **30**, 1704403.
- 146 C. Wang, T. Yokota and T. Someya, *Chem. Rev.*, 2021, **121**, 2109–2146.
- 147 K. Jeong, S. Park, G. Y. Jung, S. H. Kim, Y. H. Lee, S. K. Kwak and S. Y. Lee, *J. Am. Chem. Soc.*, 2019, **141**, 5880–5885.
- 148 X. Liu, J. Liu, S. Lin and X. Zhao, *Mater. Today*, 2020, **36**, 102–124.
- 149 B. L. Ekerdt, C. M. Fuentes, Y. Lei, M. M. Adil, A. Ramasubramanian, R. A. Segalman and D. V. Schaffer, *Adv. Healthcare Mater.*, 2018, **7**, 1800225.
- 150 H. Zhan and D. W. Löwik, *Adv. Funct. Mater.*, 2019, **29**, 1808505.
- 151 D. Fan, U. Staufer and A. Accardo, *Bioengineering*, 2019, **6**, 113.
- 152 W. Sun, H. Jiang, X. Wu, Z. Xu, C. Yao, J. Wang, M. Qin, Q. Jiang, W. Wang and D. Shi, *Nano Res.*, 2019, **12**, 115–119.
- 153 Z. Sun, C. Song, C. Wang, Y. Hu and J. Wu, *Mol. Pharm.*, 2019, **17**, 373–391.
- 154 H. Jung, M. K. Kim, J. Y. Lee, S. W. Choi and J. Kim, *Adv. Funct. Mater.*, 2020, **30**, 2004407.
- 155 C. A. Dreiss, *Curr. Opin. Colloid Interface Sci.*, 2020, **48**, 1–17.
- 156 D. Han, Y. Li, X. Liu, B. Li, Y. Han, Y. Zheng, K. W. K. Yeung, C. Li, Z. Cui and Y. Liang, *Chem. Eng. J.*, 2020, **396**, 125194.
- 157 A. Hasan, A. Khattab, M. A. Islam, K. A. Hweij, J. Zeitouny, R. Waters, M. Sayegh, M. M. Hossain and A. Paul, *Adv. Sci.*, 2015, **2**, 1500122.
- 158 Y. M. Kim, H. H. Park, D. H. Hwang, Y. Cui, E. M. Lee, S. Yahn, J. K. Lee, S. C. Song and B. G. Kim, *Nat. Commun.*, 2017, **8**, 1–14.
- 159 G. An, F. Guo, X. Liu, Z. Wang, Y. Zhu, Y. Fan, C. Xuan, Y. Li, H. Wu and X. Shi, *Nat. Commun.*, 2020, **11**, 1–12.
- 160 R. Chen, H. B. Chen, P. P. Xue, W. G. Yang, L. Z. Luo, M. Q. Tong, B. Zhong, H. L. Xu, Y. Z. Zhao and J. D. Yuan, *J. Mater. Chem. B*, 2021, **9**, 1107–1122.
- 161 R. Yang, G. Li, C. Zhuang, P. Yu, T. Ye, Y. Zhang, P. Shang, J. Huang, M. Cai and L. Wang, *Sci. Adv.*, 2021, **7**, eabg3816.
- 162 S. Zhao, P. Tseng, J. Grasman, Y. Wang, W. Li, B. Napier, B. Yavuz, Y. Chen, L. Howell, J. Rincon, F. G. Omenetto and D. L. Kaplan, *Adv. Mater.*, 2018, **30**, 1800598.
- 163 X. Sun, F. Yao and J. Li, *J. Mater. Chem.*, 2020, **8**, 18605–18623.
- 164 H. Banerjee, M. Suhail and H. Ren, *Biomimetics*, 2018, **3**, 15.
- 165 J. Xu, R. Jing, X. Ren and G. Gao, *J. Mater. Chem.*, 2020, **8**, 9373–9381.
- 166 X. Sun, P. Tyagi, S. Agate, M. G. McCord, L. A. Lucia and L. Pal, *Carbohydr. Polym.*, 2020, **234**, 115898.
- 167 X. Y. Wen, Z. Liu, J. Wang, X. Y. Tang, W. Wang, X. J. Ju, R. Xie and L. Y. Chu, *ACS Appl. Mater. Interfaces*, 2019, **11**, 20386–20393.

- 168 H. Sheng, X. Wang, N. Kong, W. Xi, H. Yang, X. Wu, K. Wu, C. Li, J. Hu, J. Tang, J. Zhou, S. Duan, H. Wang and Z. Suo, *Extreme Mech. Lett.*, 2019, **30**, 100510.
- 169 J. Nam, H. K. Lim, N. H. Kim, J. K. Park, E. S. Kang, Y. T. Kim, C. Heo, O. S. Lee, S. G. Kim, W. S. Yun, M. Suh and Y. H. Kim, *ACS Nano*, 2020, **14**, 664–675.
- 170 W. C. Huang, X. C. Ong, I. S. Kwon, C. Gopinath, L. E. Fisher, H. Wu, G. K. Fedder, R. A. Gaunt and C. J. Bettinger, *Adv. Funct. Mater.*, 2018, **28**, 1801059.
- 171 J. Liu, S. Qu, Z. Suo and W. Yang, *Natl. Sci. Rev.*, 2021, **8**, nwaa254.
- 172 N. Park and J. Kim, *Adv. Intell. Syst.*, 2020, **2**, 1900135.
- 173 Z. Lei, Q. Wang, S. Sun, W. Zhu and P. Wu, *Adv. Mater.*, 2017, **29**, 1700321.
- 174 R. S. Hsu, P. Y. Chen, J. H. Fang, Y. Y. Chen, C. W. Chang, Y. J. Lu and S. H. Hu, *Adv. Sci.*, 2019, **6**, 1900520.
- 175 C. Yang and Z. Suo, *Nat. Rev. Mater.*, 2018, **3**, 125–142.
- 176 C. H. Yang, B. Chen, J. J. Lu, J. H. Yang, J. Zhou, Y. M. Chen and Z. Suo, *Extreme Mech. Lett.*, 2015, **3**, 59–65.
- 177 Y. Ye, Y. Zhang, Y. Chen, X. Han and F. Jiang, *Adv. Funct. Mater.*, 2020, **30**, 2003430.
- 178 Y. Liang, L. Ye, X. Sun, Q. Lv and H. Liang, *ACS Appl. Mater. Interfaces*, 2019, **12**, 1577–1587.
- 179 Z. Jia, Y. Zeng, P. Tang, D. Gan, W. Xing, Y. Hou, K. Wang, C. Xie and X. Lu, *Chem. Mater.*, 2019, **31**, 5625–5632.
- 180 S. J. Devaki, R. K. Narayanan and S. Sarojam, *Mater. Lett.*, 2014, **116**, 135–138.
- 181 Y. Zhou, C. Wan, Y. Yang, H. Yang, S. Wang, Z. Dai, K. Ji, H. Jiang, X. Chen and Y. Long, *Adv. Funct. Mater.*, 2019, **29**, 1806220.
- 182 H. Sheng, X. Wang, N. Kong, W. Xi, H. Yang, X. Wu, K. Wu, C. Li, J. Hu and J. Tang, *Extreme Mech. Lett.*, 2019, **30**, 100510.
- 183 C. Keplinger, J. Y. Sun, C. C. Foo, P. Rothmund, G. M. Whitesides and Z. Suo, *Science*, 2013, **341**, 984–987.
- 184 K. Nagamine, S. Chihara, H. Kai, H. Kaji and M. Nishizawa, *Sens. Actuators, B*, 2016, **237**, 49–53.
- 185 J. Y. Sun, C. Keplinger, G. M. Whitesides and Z. Suo, *Adv. Mater.*, 2014, **26**, 7608–7614.
- 186 K. C. Spencer, J. C. Sy, K. B. Ramadi, A. M. Graybiel, R. Langer and M. J. Cima, *Sci. Rep.*, 2017, **7**, 1952.
- 187 H. Wu, G. Yang, K. Zhu, S. Liu, W. Guo, Z. Jiang and Z. Li, *Adv. Sci.*, 2021, **8**, 2001938.
- 188 L. Chen, Z. Wang, Z. Zhan, M. Xie, G. Duan, P. Cheng, Y. Chen and H. Duan, *Mater. Today Phys.*, 2021, **19**, 100404.
- 189 Q. Zhang, X. Liu, L. Duan and G. Gao, *J. Mater. Chem.*, 2020, **8**, 4515–4523.
- 190 J. Y. Sun, C. Keplinger, G. M. Whitesides and Z. Suo, *Adv. Mater.*, 2014, **26**, 7608–7614.
- 191 K. Dong, X. Liu, H. Dong, X. Zhang and S. Zhang, *Chem. Rev.*, 2017, **117**, 6636–6695.
- 192 D. Zhang, H. Qiao, W. Fan, K. Zhang, Y. Xia and K. Sui, *Mater. Today Phys.*, 2020, **15**, 100246.
- 193 G. Li, Z. Deng, M. Cai, K. Huang, M. Guo, P. Zhang, X. Hou, Y. Zhang, Y. Wang, Y. Wang, X. Wu and C. F. Guo, *npj Flexible Electron.*, 2021, **5**, 23.
- 194 Z. Wu, X. Yang and J. Wu, *ACS Appl. Mater. Interfaces*, 2021, **13**, 2128–2144.
- 195 X. J. Zha, S. T. Zhang, J. H. Pu, X. Zhao, K. Ke, R. Y. Bao, L. Bai, Z. Y. Liu, M. B. Yang and W. Yang, *ACS Appl. Mater. Interfaces*, 2020, **12**, 23514–23522.
- 196 J. L. Li, R. Y. Wang, X. Y. Liu and H. H. Pan, *J. Phys. Chem. B*, 2009, **113**, 5011–5015.
- 197 Z. Zhao, S. Zhuo, R. Fang, L. Zhang, X. Zhou, Y. Xu, J. Zhang, Z. Dong, L. Jiang and M. Liu, *Adv. Mater.*, 2018, **30**, 1804435.
- 198 J. Song, S. Chen, L. Sun, Y. Guo, L. Zhang, S. Wang, H. Xuan, Q. Guan and Z. You, *Adv. Mater.*, 2020, **32**, 1906994.
- 199 F. Chen, D. Zhou, J. Wang, T. Li, X. Zhou, T. Gan, S. Handschuh-Wang and X. Zhou, *Angew. Chem.*, 2018, **130**, 6678–6681.
- 200 S. Shi, X. Peng, T. Liu, Y. N. Chen, C. He and H. Wang, *Polymer*, 2017, **111**, 168–176.
- 201 J. Wu, Z. Wu, H. Xu, Q. Wu, C. Liu, B. R. Yang, X. Gui, X. Xie, K. Tao and Y. Shen, *Mater. Horiz.*, 2019, **6**, 595–603.
- 202 H. Gao, Z. Zhao, Y. Cai, J. Zhou, W. Hua, L. Chen, L. Wang, J. Zhang, D. Han and M. Liu, *Nat. Commun.*, 2017, **8**, 1–8.
- 203 X. Pan, Q. Wang, R. Guo, Y. Ni, K. Liu, X. Ouyang, L. Chen, L. Huang, S. Cao and M. Xie, *J. Mater. Chem.*, 2019, **7**, 4525–4535.
- 204 W. Xie, J. Duan, H. Wang, J. Li, R. Liu, B. Yu, S. Liu and J. Zhou, *J. Mater. Chem.*, 2018, **6**, 24114–24119.
- 205 J. Luo, Y. Xing, C. Sun, L. Fan, H. Shi, Q. Zhang, Y. Li, C. Hou and H. Wang, *Chem. Eng. J.*, 2022, **427**, 130886.
- 206 V. Saunier, E. Flahaut, M. C. Blatché, C. Bergaud and A. Maziz, *Biosens. Bioelectron.*, 2020, **165**, 112413.
- 207 D. Qi, Z. Liu, M. Yu, Y. Liu, Y. Tang, J. Lv, Y. Li, J. Wei, B. Liedberg and Z. Yu, *Adv. Mater.*, 2015, **27**, 3145–3151.
- 208 R. Liu, S. Zhao and J. Liu, *ACS Appl. Electron. Mater.*, 2021, **3**, 101–118.
- 209 R. Green and M. R. Abidian, *Adv. Mater.*, 2015, **27**, 7620–7637.
- 210 N. A. Kotov, J. O. Winter, I. P. Clements, E. Jan, B. P. Timko, S. Campidelli, S. Pathak, A. Mazzatenta, C. M. Lieber and M. Prato, *Adv. Mater.*, 2009, **21**, 3970–4004.
- 211 W. K. Oh, O. S. Kwon and J. Jang, *Polym. Rev.*, 2013, **53**, 407–442.
- 212 S. P. Lacour, S. Benmerah, E. Tarte, J. FitzGerald, J. Serra, S. McMahon, J. Fawcett, O. Graudejus, Z. Yu and B. Morrison, *Med. Biol. Eng. Comput.*, 2010, **48**, 945–954.
- 213 L. Guo, M. Ma, N. Zhang, R. Langer and D. G. Anderson, *Adv. Mater.*, 2014, **26**, 1427–1433.
- 214 D. P. Qi, Z. Y. Liu, Y. Liu, Y. Jiang, W. R. Leow, M. Pal, S. W. Pan, H. Yang, Y. Wang, X. Q. Zhang, J. C. Yu,

- B. Li, Z. Yu, W. Wang and X. D. Chen, *Adv. Mater.*, 2017, **29**, 10.
- 215 D. Qi, Z. Liu, Y. Liu, Y. Jiang, W. R. Leow, M. Pal, S. Pan, H. Yang, Y. Wang, X. Zhang, J. Yu, B. Li, Z. Yu, W. Wang and X. Chen, *Adv. Mater.*, 2017, **29**, 1702800.
- 216 E. Cuttaz, J. Goding, C. Vallejo-Giraldo, U. Aregueta-Robles, N. Lovell, D. Ghezzi and R. A. Green, *Biomater. Sci.*, 2019, **7**, 1372–1385.
- 217 J. Che, Y. Xiao, X. Zhu and X. Sun, *Polym. Int.*, 2008, **57**, 750–755.
- 218 N. K. Guimard, N. Gomez and C. E. Schmidt, *Prog. Polym. Sci.*, 2007, **32**, 876–921.
- 219 K. A. Ludwig, J. D. Uram, J. Yang, D. C. Martin and D. R. Kipke, *J. Neural Eng.*, 2006, **3**, 59.
- 220 A. R. Harris, S. J. Morgan, J. Chen, R. M. Kapsa, G. G. Wallace and A. G. Paolini, *J. Neural Eng.*, 2012, **10**, 016004.
- 221 R. A. Green, N. H. Lovell and L. A. Poole Warren, *Acta Biomater.*, 2010, **6**, 63–71.
- 222 M. Cucchi, C. Gruener, L. Petrauskas, P. Steiner, H. Tseng, A. Fischer, B. Penkovsky, C. Matthus, P. Birkholz and H. Kleemann, *Sci. Adv.*, 2021, **7**, eabh0693.
- 223 S. Guzzo, S. Carli, B. Pavan, A. Lunghi, M. Murgia and M. Bianchi, *Nanomaterials*, 2021, **11**, 2022.
- 224 J. Ouyang, *SmartMat*, 2021, **2**, 263–285.
- 225 C. Wang, K. He, J. Li and X. Chen, *SmartMat*, 2021, **2**, 252–262.
- 226 A. Blau, A. Murr, S. Wolff, E. Sernagor, P. Medini, G. Iurilli, C. Ziegler and F. Benfenati, *Biomaterials*, 2011, **32**, 1778–1786.
- 227 E. A. Cuttaz, C. A. R. Chapman, O. Syed, J. A. Goding and R. A. Green, *Adv. Sci.*, 2021, **8**, 2004033.
- 228 L. P. Wang, W. Wang, L. Di, Y. N. Lu and J. Y. Wang, *Colloids Surf., B*, 2010, **80**, 72–78.
- 229 M. R. Abidian and D. C. Martin, *Biomaterials*, 2008, **29**, 1273–1283.
- 230 M. R. Abidian, J. M. Corey, D. R. Kipke and D. C. Martin, *Small*, 2010, **6**, 421–429.
- 231 P. Cai, C. Wan, L. Pan, N. Matsuhisa, K. He, Z. Cui, W. Zhang, C. Li, J. Wang and J. Yu, *Nat. Commun.*, 2020, **11**, 1–12.
- 232 C. M. Tringides, N. Vachicouras, I. de Lázaro, H. Wang, A. Trouillet, B. R. Seo, A. Elosegui-Artola, F. Fallegger, Y. Shin and C. Casiraghi, *Nat. Nanotechnol.*, 2021, 1–11.
- 233 N. Chen, B. Luo, A. C. Patil, J. Wang, G. G. L. Gammad, Z. Yi, X. Liu, S. C. Yen, S. Ramakrishna and N. V. Thakor, *ACS Nano*, 2020, **14**, 8059–8073.
- 234 J. Goding, C. Vallejo-Giraldo, O. Syed and R. Green, *J. Mater. Chem. B*, 2019, **7**, 1625–1636.
- 235 J. Chung, A. Khot, B. M. Savoie and B. W. Boudouris, *ACS Macro Lett.*, 2020, **9**, 646–655.
- 236 F. Fu, J. Wang, H. Zeng and J. Yu, *ACS Mater. Lett.*, 2020, **2**, 1287–1301.
- 237 S. Yang, L. Jang, S. Kim, J. Yang, K. Yang, S. W. Cho and J. Y. Lee, *Macromol. Biosci.*, 2016, **16**, 1653–1661.
- 238 A. R. Spencer, A. Primbetova, A. N. Koppes, R. A. Koppes, H. Fenniri and N. Annabi, *ACS Biomater. Sci. Eng.*, 2018, **4**, 1558–1567.