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Unveiling lasing mechanism in CsPbBr₃ microsphere cavities†

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Recently, the light–matter interaction of perovskite microcavities has been widely explored for its great potential in low-threshold lasing devices. However, the mechanism of perovskite lasing remains unclear to date. In this study, we demonstrated high-quality single-mode excitonic lasing in CsPbBr₃ microspheres, providing an ideal platform to study the underlying physics of lasing behavior. We show that the lasing mechanism shifts from the exciton–exciton scattering to the exciton–phonon scattering with the increase in temperature from 77 to 300 K, which was verified by temperature-dependent photoluminescence (PL), time-resolved photoluminescence (TRPL) as well as temperature-dependent Raman spectroscopy. Furthermore, by analyzing PL line width broadening with varied temperatures, we found that two different phonon modes were involved in the exciton–phonon scattering process. The scattering from the low-energy phonon (~8.6 meV) is the dominant source of exciton–phonon coupling in the intermediate temperature range (77 to 230 K), while the high-energy phonon (~15.3 meV) dominates from 230 K to room temperature. These results confirm the lasing mechanism in such perovskite-based micro/nano-cavities and significantly influence the development of future low-threshold lasers.

1. Introduction

Small solid-state lasers have shown great potential in applications such as optical integration, high-speed communication

and high-resolution imaging. Over the past decades, a variety of semiconductor nanostructures including nanowires (NWs), nanoribbons, and quantum dots (QDs) have been utilized to serve as active materials for small lasers.^{1–8} Despite these great successes, high lasing threshold is still an obstacle in small lasers due to their large optical losses.^{9–11} Recently, lead halide perovskites have attracted considerable attention for their unprecedented success in the field of photovoltaics research^{12–15} due to their high absorption coefficient, large carrier diffusion length and low density of defects.^{16–18} Moreover, perovskites provide excellent gain, which is crucial for a high-performance coherent light source.^{19–21} Till now, perovskite-based small lasers have been realized in many micro/nanostructures at room temperature, such as NWs,^{22–27} nanoplatelets (NPs),^{28,29} and QDs.^{30–32} Compared to the conventional II–VI/III–V semiconductors, the optically pumped perovskite microlasers based on such nanostructures show even lower threshold, better spectral coherence, and wide-range emission colours.^{21,24,25,33,34}

However, it remains unclear whether the emission of perovskites originates from free charge carriers or excitons,^{34–38} and the mechanism of lasing in these microcavities has been rarely discussed, although it is critical to the rational development of small lasers. The basic condition of lasing is that the stimulated emission is greater than the absorption, that is, the optical gain is positive. Previously, two main mechanisms were generally involved to produce significant gain: excitonic and electron–hole plasma (EHP) process.^{39–42} In the intermediate density regime where excitons are still good quasiparticles, the following stimulated emission or lasing due to various excitonic processes was observed: (1) exciton–exciton (ex–ex) scattering; (2) exciton-longitudinal optical phonon (ex–LO) scattering; and (3) exciton–electron (ex–el) scattering. However, excitons are no longer good quasi particles in the high density regime, which means that excitonic processes are hardly relevant at these densities. The laser action is converted from an excitonic process to an EHP emission. An EHP state is formed when the density of electron–hole pairs exceeds the Mott

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density n_M , where screening effect sufficiently reduces the Coulomb interaction such that bound excitonic states cannot exist.^{38,43} Despite these intrinsic properties, the geometry and quality of microcavities and excitation methods can also affect the optical gain and lasing mechanism.^{40,42} Temperature-dependent spontaneous emission (SE) and lasing spectroscopy are well-established methods to study the lasing mechanism. However, the multimode lasers inevitably introduce big errors. Very recently, the low threshold, single-mode lasing has been demonstrated in microsphere/nanocuboid cavities, which provide an ideal system to explore the light–matter interaction involved in the lasing process.^{38,44,45} However, the underlying physics of a particular lasing behavior has barely been researched, and the lasing mechanism in these CsPbBr₃ microspheres remains unclear and controversial to date. Accordingly, some researchers thought that the emission of perovskites originated from excitons,³⁴ while others considered it to originate from the bimolecular recombination of free charge carriers.³⁷ Yang *et al.* postulated that an EHP mechanism is responsible for the stimulated emission in CsPbBr₃ nanowires at the carrier density three times larger than the bulk CsPbBr₃ Mott density.²² Later, Wang *et al.* and Liu *et al.* proposed that the formation of EHP was the origin of the contribution from stimulated emission to optical gain in CsPbBr₃ nanostructures under high-intensity two-photon excitation.^{38,46}

In this study, we performed temperature-dependent photoluminescence (PL) and time-resolved photoluminescence (TRPL) experiments as well as Raman spectroscopy to confirm the mechanism of lasing in CsPbBr₃ microsphere cavities. When the temperature T is 77–150 K, the lasing mechanism is dominated by ex–ex scattering, while the exciton–phonon scattering plays a key role when T is 150–300 K. Systemic Raman and temperature-dependent SE spectroscopy suggest that two phonon modes (~ 8.6 meV and ~ 15.3 meV) were involved in the exciton–phonon scattering process by analyzing the PL line width at different temperature regions. The scattering from the low-energy phonon is the dominant source of exciton–phonon coupling in the intermediate temperature range (77 to 230 K), while the high-energy phonon dominates from 230 K to the room temperature. All of our experimental results suggest that the lasing mechanism is ascribed to the formation of excitons (scattering with exciton or phonon) instead of electron–hole plasma at all temperature regions.

2. Results and discussion

CsPbBr₃ microspheres were grown on silicon wafer substrates by the chemical vapor deposition (CVD) method (for growth conditions, see ESI, Section I†). The microsphere structures were selected for the following two reasons: (1) the spherical structure can support the whispering gallery mode (WGM) with relatively higher Q -factor compared with the other Fabry–Pérot (FP) or WGM microcavities confined in NWs and NPs.^{44,45} Hence, the lasing could be realized at a quite low

pumping intensity in order to suppress the heating, band filling, band renormalization and screening effects occurring at high excitation density. (2) The diameter of CsPbBr₃ microspheres is as small as several microns; with this dimension NWs or platelets can be replaced by microspheres for single mode lasers; therefore, the temperature-dependent lasing wavelength and emission width can be precisely resolved. Fig. 1a shows a typical SEM image of CsPbBr₃ microspheres grown on a silicon substrate with diameters varying from 1.0 to 3.0 μm . They have a perfectly spherical configuration and good smooth surface (inset, Fig. 1a), which can be clearly verified further in the SEM images of the sphere samples with different tilted angles (see ESI, Section II†). Energy-dispersive spectrometer (EDS) mapping confirms the uniform spatial distribution of the contained chemical elements with a clear boundary of the CsPbBr₃ microsphere (Fig. 1b). The X-ray diffraction (XRD) pattern (Fig. 1c) shows the two peaks corresponding to the (002) and (110) face of the orthorhombic phase of CsPbBr₃,²² which is quite stable below 370 K.^{47,48} As shown in Fig. 1d, the CsPbBr₃ microsphere presents a broad absorption band with the edge at ~ 530 nm. The PL peak is located at 532 nm at room temperature with full width at half maximum (FWHM) of ~ 20 nm, which is consistent with previous researches.³⁵ These characterizations of morphology and spectroscopy confirmed the high quality of the as-grown CsPbBr₃ microspheres.

Room-temperature optically pumped PL spectroscopy was performed on the CsPbBr₃ microspheres. The schematic is



Fig. 1 Characterizations of CsPbBr₃ microspheres. (a) A typical SEM image of the CsPbBr₃ microcrystals grown on silicon wafer showing a sphere shape. Scale bar is 5 μm . Inset is the magnified SEM image of a typical CsPbBr₃ microsphere. Scale bar is 0.5 μm . (b) EDS mapping of CsPbBr₃ microsphere, showing the uniform composition of the CsPbBr₃ microspheres. The SEM image is shown for comparison. (c) XRD pattern of the as-grown CsPbBr₃ microspheres. (d) Optical absorption spectrum (navy) and PL spectra (olive) of the sample of CsPbBr₃ microspheres, and insert shows the photoluminescence image of the CsPbBr₃ microspheres. The scale bar inside is 5 μm .



Fig. 2 Coherent light emission from the CsPbBr₃ microspheres. (a) Schematic of the experimental setup, a CsPbBr₃ microsphere lying on a SiO₂ substrate. A 400 nm (80 fs, 1 kHz) pulsed laser source (violet arrow) pumping the entire microsphere, and the microsphere emitting in green. (b) Emission spectra of CsPbBr₃ microspheres with increasing pump fluence around threshold, showing the broadening emission spectra before threshold, and a sudden sharp laser peak at and above threshold. (c) Output intensity as a function of excitation power. Black lines are fitting curves with $I = \sim P^k$. Insets: Optical images (with pump laser filtered out) showing emission from the CsPbBr₃ microsphere below threshold (left), and above threshold (right). The scale bar is 2 μm . (d) For spontaneous emission and lasing, the wavelength and line width versus the excitation pump fluence. (e) Single-mode lasing spectra of eight typical CsPbBr₃ microspheres with different diameters. Inset: Magnified SEM image of CsPbBr₃ microspheres with diameter from ~ 1 to 3 μm . The scale bar is 1 μm . (f) The resonant mode is extracted and plotted as a function of the diameter of CsPbBr₃ microspheres. Upper left inset: Simulated electric field distribution under a transverse magnetic resonant mode at ~ 540 nm. Bottom inset: Amplified spectrum for CsPbBr₃ microsphere. The experimental data (green dots) was well fitted by Lorentz curve, giving an ultra-narrow FWHM of ~ 0.15 nm and high-quality factor Q of ~ 3600 .

shown in Fig. 2a, where a 400 nm-fs-pulse laser pumps the whole body of CsPbBr₃ microspheres to ensure high excitation efficiency and low heating effects. The green line illuminates the propagation mode of light in individual microsphere *via* total reflection on the microsphere-air interface. The pump-fluence-dependent emission spectra in Fig. 2b show the transition from SE to laser behavior in a typical CsPbBr₃ microsphere with a diameter of ~ 2 μm . At low pump fluence, there is only a single broad emission peak located at ~ 530 nm owing to SE. When pump fluence exceeds 3.5 $\mu\text{J cm}^{-2}$, one sharp peak generates above the SE profile and further becomes dominant in the emission spectra. The relation of integrated intensity *versus* pump fluence in Fig. 2c reveals an S-shaped curve, confirming the achievement of lasing with a threshold of 3.5 $\mu\text{J cm}^{-2}$. The transition can also be indicated from the optical image at different pump fluence. Below the threshold, the microsphere mainly exhibits uniform luminescence, which is maintained by the laser spot (left inset, Fig. 2c). Coherent light emission can be inferred from the optical image above the threshold (right inset, Fig. 2c) with a multi-circular pattern originated from the interference in the WGM microcavity. Furthermore, the saturation region (>5 $\mu\text{J cm}^{-2}$) has the same slope with the spontaneous emission region fitted by $I = P^k$. The value of $1 < k < 2$ is just consistent with excitonic emission for excitation laser energy exceeding the gap energy.⁴⁹ The carrier density at the threshold is estimated to be $\approx 7.0 \times 10^{16}$ cm^{-3}

under the present excitation conditions, which is below the Mott density of CsPbBr₃ reported (from 1.8×10^{17} to 1.0×10^{19} cm^{-3}),^{21,50} suggesting that the lasing may not be due to EHP lasing.^{21,40,41} When the pump intensity is above the threshold, the position and FWHM of the lasing mode show a non-monotonic shift or a slight change (Fig. 2d), also suggesting that the laser emission occurring in the CsPbBr₃ microspheres should be an excitonic property rather than an EHP emission.⁴⁰ Moreover, the large exciton binding energy (from 65–86 meV, ESI, Section III†) and approximate linear dependence of PL intensity on laser power (see ESI, Section IV†) also verify that excitons instead of free charge carriers exist at room temperature and at these excited conditions.

The observed lasing emission in these microspheres is usually single mode and on the lower energy side of the PL peak, as is illustrated in Fig. 2e. The generation of low-energy laser modes can be explained as the counterbalance of optical gain and finite self-absorption at absorption tail.⁵¹ The laser peak can be modulated from 537 to 542 nm by changing the diameter of the microsphere from 1 to 3 μm . This can be explained by the relation of the WGM and cavity length in a spherical cavity,⁴⁴

$$\pi D/(\lambda/N) \approx \nu. \quad (1)$$

Here, the wavelength of the mode peak λ at different mode orders ν is determined by the optical path of the spherical

cavity with the parameters of diameter D and refractive index of the active material N . For a fixed mode order, the resonant WGM peak scales linearly with the diameter of the spherical cavity. This is consistent with the relation of single-mode laser peak *versus* diameter of CsPbBr₃ microspheres, as depicted in Fig. 2f. When the diameter of CsPbBr₃ microspheres increases, lasing peaks shift to longer wavelength, which can be fitted linearly as expected from eqn (1). To investigate the mode type of the single-mode laser in this spherical WGM microcavity, finite element simulation by FDTD solution was employed with consideration of the material dispersion of CsPbBr₃. Upper-left inset of Fig. 2f shows the electric field distribution of transverse magnetic (TM) polarized WGM modes (at wavelength of ~ 540 nm) in the microsphere with the diameter of 2.48 μm , which is consistent with the experimental results (see inset of Fig. 2c). The preferred TM-mode lasing in these CsPbBr₃ microspheres can also be seen in CsPbBr₃ perovskite nanoplatelets and microdisks.^{34,52} In addition, a sharp lasing peak presented in the lower-right inset of Fig. 2f indicates the high-quality laser mode. Using the Lorentz function fitting, a FWHM of 0.15 nm is obtained. The quality factor Q is further determined to be ~ 3600 by $Q = \lambda/\delta\lambda$, where λ and $\delta\lambda$ are the center wavelength and FWHM of the peak profile, respectively.

These values in such sub-micron spheres are better than those in other nanolasers reported in previous literatures.^{22,38,53} In general, the Q factor can be degraded in the smaller cavity approaching the wavelength of light due to the increased leakage of light. The excellent lasing performance in these CsPbBr₃ microspheres can be understood to be due to the high-quality single crystalline characteristic with a smooth surface avoiding radiation loss in the corners and low defects, together with the high optical gain in CsPbBr₃.

Next, temperature-dependent lasing behavior was investigated to explore the mechanism of lasing in the CsPbBr₃ microspheres. Fig. 3a presents the normalized lasing spectra of a typical CsPbBr₃ microsphere when the temperature varies from 77 to 300 K. When the temperature is in the range of 77–150 K, the lasing position locates at the low-energy side of SE and almost remains unchanged; as the temperature increases from 150 to 300 K, the lasing position redshifts from 533.4 nm to 535.2 nm, which is consistent with reports from previous literature on CsPbBr₃ quantum dots.^{39,54} The variation tendency of the single-mode lasing peak with the temperature is consistent with that of the PL peak position, as presented in Fig. 3b. A summary of these temperature-dependent data in Fig. 3c clearly shows the weaker temperature depen-

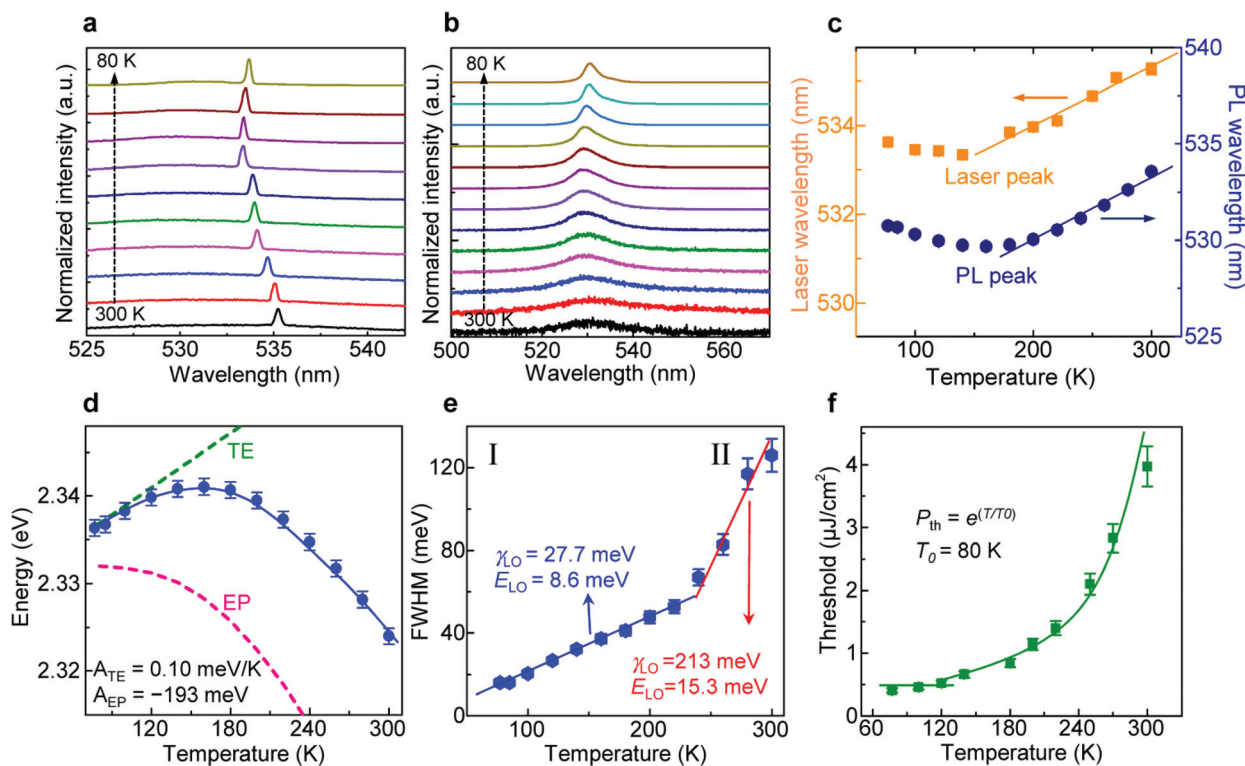


Fig. 3 Temperature-dependent photoluminescence of the CsPbBr₃ microspheres. (a) Normalized lasing spectra of a typical CsPbBr₃ microsphere at different temperatures from 77 to 300 K. (b) Normalized PL spectra of a typical CsPbBr₃ microsphere at a temperature range from 77 to 300 K. (c) A summary of the variation tendency of the single-mode lasing peak and PL emission wavelength with temperature. (d) The emission peak energy of the microspheres with increasing temperatures from 77 to 300 K. The solid line is the fitting result; the dashed lines show the individual contributions of TE and EP interaction. (e) Plot of the change in FWHM with temperature from 77 K to 300 K, presenting a drastic change at 235 K. The solid line is the fit according to eqn (2). (f) Temperature-dependent lasing threshold in the range from 77 to 300 K, fitted by the exponential function, giving a characteristic temperature $T_0 = 80$ K.

dence of the spontaneous and stimulated emission peaks below 150 K, and the following monotonic red-shift at the slope of ~ 0.04 nm K^{-1} at higher temperatures. This consistency indicates that the behaviors of the single-mode lasing have been derived from the material gain with temperature rather than optical cavity mode gain modulation, as shown in Fig. 2e.

A detailed analysis of the PL spectra under different temperatures ranging from 77 to 300 K was conducted in order to confirm the lasing mechanism of CsPbBr₃ microspheres supposed above. By assuming a linear relationship between lattice constant and temperature, the temperature dependence of the PL peak energy is generally estimated by the following expression,⁵⁴

$$E(T) = E_0 + A_{TE}T + A_{EP} \left(\frac{2}{\exp(\hbar\omega/k_B T) - 1} + 1 \right) \quad (2)$$

where E_0 is the unrenormalized bandgap, A_{TE} and A_{EP} are the weight of the thermal expansion (TE) and exciton–phonon (EP) interaction, respectively, and $\hbar\omega$ is the average optical phonon energy. By fitting the temperature-dependent PL peak with eqn (2) (the blue line in Fig. 3d), the following parameters are obtained: $E_0 = 2.52$ eV, $A_{TE} = 0.10$ meV K^{-1} , and $A_{EP} = -193$ meV. By substituting the above fitted parameters into eqn (1), we plot the TE term and the EP interaction term separately, as the dashed lines indicated in Fig. 3d. It clearly shows that the little temperature-dependent PL peak below 150 K can be derived from the joint contribution of TE and EP. However, optical phonon modes are significantly populated at higher temperatures, leading to an increasingly negative contribution. As a result, the increasing rate of the PL peak energy diminished, even reversed from 160 to 300 K. The energy of the PL peak is expected to further decrease at temperatures higher than 300 K, indicating that no other factors contribute to the PL peak energy evolution. This strongly suggests that the ex-phonon process is responsible for lasing in this temperature range and able to survive up to room temperature.

Furthermore, we can see an inflection point near 230 K in the plot of the change in FWHM with the temperature (Fig. 3e), where the slope of the curve from 77 to 230 K is different from that above 230 K. The FWHM of the PL measured above 230 K increases faster than that below 230 K. The analysis of temperature-dependent emission broadening has long been used to evaluate the mechanism of exciton–phonon coupling in a large number of inorganic semiconductors. The emission broadening can be expressed as⁵⁵

$$\Gamma(T) = \Gamma_0 + \gamma_{LO} \left(\frac{E_{LO}}{e^{\hbar\omega/k_B T} - 1} \right)^{-1}, \quad (3)$$

where Γ_0 is the term of a temperature-independent inhomogeneous broadening, which results from the disorder and imperfection scattering. The second term is the homogeneous broadening term, which arises from LO phonon–charge scattering. γ_{LO} and E_{LO} are carrier–phonon coupling strength and the LO phonon energy, respectively. Piecewise fitting of the temperature-dependent PL line width with temperature using

eqn (3) was plotted in Fig. 3e, yielding the coupling strength $\gamma_{LO} = 27.7$ meV and an average phonon energy of $E_{LO} = 8.6$ meV in region I (below 230 K), and the coupling strength $\gamma_{LO} = 213$ meV and an average phonon energy of $E_{LO} = 15.3$ meV in region II (from 230 to 300 K). Therefore, a preliminary conclusion can be drawn that phonons of different energies are dominant at different temperature regions, and the exciton–phonon coupling strength γ_{LO} in region II is much stronger than that in region I. Thus, we conjecture that two different phonon modes participate in the exciton–phonon scattering, and the ex-LO phonon scattering that involves the phonon with energy of $E_{LO} = 15.3$ meV is more significant.

In addition, the lasing threshold (P_{th}) can be increased from ~ 0.4 μJ cm^{-2} to 3.5 μJ cm^{-2} when the sample temperature rises from 77 K to 300 K (Fig. 3f). Therefore, the threshold dependence shows little temperature dependence at lower temperature accompanied by nearly constant SE and lasing position, which is consistent with the reported lasing from exciton–exciton scattering.⁴² At temperatures above 150 K, the higher exciton states become thermally populated and cause an increase in the lasing threshold of the exciton–exciton process. Exciton scattering with LO phonons can contribute more to gain at these temperatures at the same time, which leads to the threshold rising exponentially due to the thermal population of the optical phonon branch. Therefore, at the higher temperature, the lasing mechanism of CsPbBr₃ microspheres can be attributed to the ex-LO phonon scattering process. Moreover, this temperature-dependent variation tendency is fitted by the empirical function $P_{th} = \sim e^{T/T_0}$, where T is the measured temperature and T_0 is a characteristic temperature. A large T_0 indicates a weak dependence on the temperature of P_{th} , which suggests that a semiconductor laser has good quality. Here, the obtained T_0 is 80 K in the temperature range of 77–300 K, which is comparable to that of a conventional semiconductor.^{56,57}

To confirm the exciton–phonon coupling in sphere microcavities, Raman measurement was executed using 633 nm-laser as the excitation source (see ESI, Section V†). There is no effect of phase change on the temperature-dependent Raman spectra from 103 K to 303 K, as mentioned above. As it can be observed from Fig. 4a recorded at $T = 103$ K, Raman spectrum of the CsPbBr₃ perovskite microsphere has two major regions: one region near 70 cm^{-1} with three sharp major peaks, and another region near 130 cm^{-1} with broad, overlapping peaks. Fitted by Lorentz function to obtain the exact value of wave number, five Raman modes centered at around 60, 70, 76, 127 and 141 cm^{-1} were, respectively, recognized (as labeled in Fig. 4a), which agree well with a previous report.⁵⁸ A detailed analysis of the vibrational modes of the CsPbBr₃ microsphere reveals that they are derived mainly from the $[PbBr_6]^{4-}$ anion. Based on previous work,⁵⁹ the peak at 141 cm^{-1} is assigned to the symmetric Pb–Br stretching mode, whereas the peak at 127 cm^{-1} is the antisymmetric partner. The strong modes at 76 cm^{-1} and 70 cm^{-1} are assigned to the bending mode of Br–Pb–Br and its antisymmetric partner. But the peak at 62 cm^{-1} is indeterminate, it may be assigned to Pb–Br bending or lattice deformation mode, and needs to be further

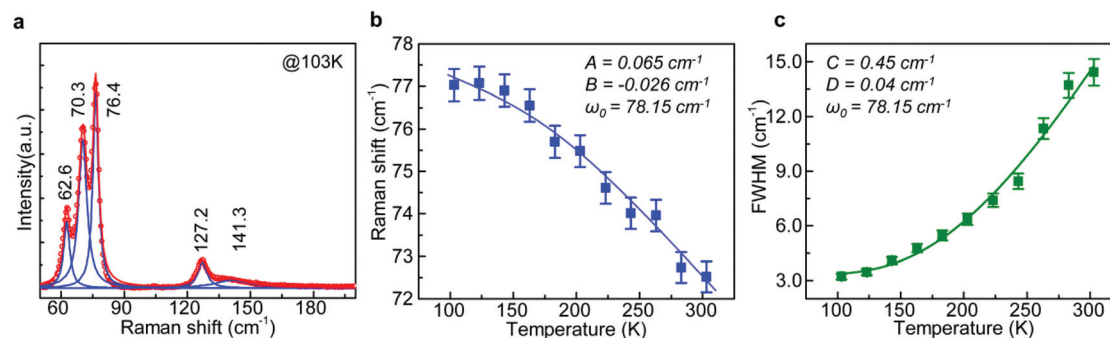


Fig. 4 Temperature-dependent Raman spectra of the CsPbBr₃ microspheres. (a) Raman spectrum of the CsPbBr₃ microspheres measured with a 633 nm excitation laser at 103 K. Five Raman peaks at 62.6, 70.3, 76.4, 127.2 and 126 cm⁻¹ are obtained by peak fitting. (b) and (c) Temperature dependence of position and line width of the Raman mode at 76 cm⁻¹. Inside are fitting parameters using the equations from ref. 54.

studied. Nevertheless, Raman modes at ~ 70 cm⁻¹ and ~ 127 cm⁻¹ are regarded as the signature of phonons with energy of 8.6 meV fitted in region I of Fig. 3e and of 15.3 meV obtained in region II of Fig. 3e, respectively. As mentioned earlier, no structural phase transitions were observed below 370 K. Hence, the different scattering mechanisms are derived from the participation of different vibrational modes, in other words, coupling with different LO phonons. The 76 cm⁻¹ phonon mode plays a key role but has a weaker coupling strength in region I, and in region II the 126 cm⁻¹ phonon mode is dominant with stronger coupling strength with the increase in temperature. To further analyze the temperature-dependent Raman spectra, we extracted the Raman shift and FWHM of the main peak at 76 cm⁻¹ to plot the change with temperature (Fig. 4b and c). Without structural phase transitions, the major contribution to the temperature dependence of the phonon wavenumber is expected to be the anharmonicity. In this model, the variation of phonon frequency $\omega(T)$ and width $\Gamma(T)$ can be described by an anharmonic expression⁶⁰ (see ESI, Section VI†). As shown in Fig. 4b and c, the experimental results for Raman frequency show blue shift and the line-width broadens as the temperature increases from

103 K to 303 K. Also shown is the fit to the data (solid curve) specified by the above model. The experimental phenomena of line width and wave number conform to the anharmonic behavior,⁶¹ with A , B , C and D of 0.065 cm⁻¹, -0.026 cm⁻¹, 0.45 cm⁻¹ and 0.04 cm⁻¹, respectively, and ω_0 of 78.15 cm⁻¹. The anharmonic behavior indicates that there is an interaction between phonons,^{62,63} also known as thermal expansion, which is consistent with the fitting results of the temperature dependence of the PL peak energy in Fig. 3d.

Temperature-dependent TRPL spectra were measured to study the exciton recombination dynamics of the CsPbBr₃ microsphere. Fig. 5a plots three representative PL decay curves at 77 K, 180 K and 280 K. All experimental data and multi-exponential decay fitting results are presented in the ESI (see Section VII†). The average lifetime provides an overall comparison (including all the recombination processes) of the dynamic behavior.^{54,64} We summarize the average lifetime and the integrated PL intensity at different temperatures in Fig. 5b and c. It is found that they share the same trend including a turning point, which is similar to the changing of PL FWHM with temperature. When the temperature increases from 77 K to 230 K, the average lifetime shortens from ~ 4 ns to ~ 1.7 ns as shown in

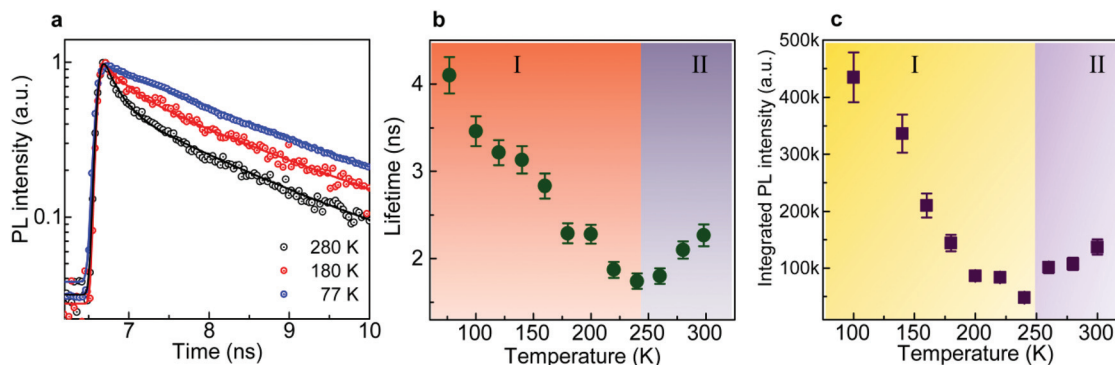


Fig. 5 Time-resolved PL (TRPL) of CsPbBr₃ microspheres. (a) TRPL decay curves of CsPbBr₃ microspheres with different temperatures monitored around 530 nm with excitation of 400 nm (76 MHz, 120 fs). (b) The PL lifetime of CsPbBr₃ microspheres decreases with temperature from 77 K to 238 K, while it increases with temperature from 238 K to 300 K. (c) Plot of the change in integrated PL intensity with temperature from 77 K to 300 K. It almost shares the same feature as shown in (b).

Fig. 5b. In this scenario, the interaction between exciton and low-energy phonon (~ 8.6 meV) as well as the thermal expansion are speculated to jointly influence the lifetime in the temperature range from 77 K to 230 K, where low-energy phonons become more active. The interaction with the lattice becomes predominant and reduces the PL lifetime as carriers transfer their energy to the lattice preventing the radiative recombination. The intensity of PL is decreased with temperature as demonstrated in Fig. 5c. It is confirmed that thermal quenching, which can result in a decrease in PL lifetime,⁶⁵ is responsible for the change of PL intensity in this temperature region. Further increasing the temperature from 230 K to 300 K leads to the activation of high-energy phonons (~ 15.3 meV), which could assist the relaxation of hot carriers from higher-energy states to lower-energy states, and thus the average lifetime lengthens expectedly. The PL intensity hence turns out to be opposite to the trend at low temperature (Fig. 5c).

3 Conclusions

To conclude, the lasing mechanism in CsPbBr₃ microspheres was elucidated by temperature-dependent PL and TRPL measurements as well as Raman spectroscopy. Our results show that it is purely excitonic lasing in the CsPbBr₃ microspheres: ex-ex scattering at the low-temperature range and ex-phonon scattering at higher-temperature range. Two phonon modes of different energies (~ 8.6 meV and ~ 15.3 meV) have been observed and confirmed to participate in the exciton-phonon coupling in different temperature ranges. The scattering from the low-energy phonon is the dominant source of exciton-phonon coupling in the intermediate temperature range, while the high-energy phonon dominates from 230 K to room temperature. The different exciton-phonon coupling mechanisms accompanied by TE in turn determine the behaviors of PL decay and PL intensity with temperature variations. Although the mechanism will be affected by factors such as the crystal properties, cavity configuration, losses, and excitation conditions, we believe our research lays the groundwork for the physics underlying the lasing behavior in perovskites and also helps designing ultralow-threshold lasing devices.

4 Experimental sections

Morphology and structure characterization

The morphology of the as-grown CsPbBr₃ microspheres were characterized by scanning electron microscopy (SEM, NOVA NanoSEM 430) equipped with energy dispersive spectroscopy (EDS, Oxford: X-Max80), and the crystal structure was measured using X-ray diffraction (XRD, Rigaku D/MAX-TTRIII (CBO), Cu K α radiation).

Optical characterizations

Absorption spectra were obtained by a home-built micro-transmission/absorption spectrometer. The light emitted

from a halogen lamp source was vertically focused onto the sample from top. The reflected light from the sample was collected by a reflective objective, and spectrally analyzed by a monochromator. Apertures with adjustable size were used to selectively acquire reflected light from the targeted sample. For obtaining steady-state PL spectrum, a 405 nm-CW laser was focused onto an individual CsPbBr₃ microsphere by an Olympus BX51 microscope. The PL emission signal was collected by the same microscope objective in a backscattering configuration and analyzed by a Princeton Instrument SP2500i spectrometer equipped with a liquid nitrogen-cooled charge-coupled detector (CCD). The PL image was recorded by the CCD equipped on Olympus BX51 microscope. The PL lifetime was obtained by a time-corrected single photon counter (TCSPC) with an ultimate temporal resolution of ~ 40 ps. A frequency-doubled mode-locked Ti-sapphire oscillator laser (800 nm, repetition rate 76 MHz, pulse length 120 fs) was used as the excitation laser source. For lasing experiments, the excitation laser source was generated by a frequency-doubled crystal from a Coherent Astrella regenerative amplifier (80 fs, 1 kHz, 2.5 mJ per pulse, 800 nm) that was seeded by a Coherent Vitara-s oscillator (35 fs, 80 MHz). The setup of optical measurement is presented in the ESI, Section VIII.†

Simulation of electric field distribution of CsPbBr₃ microsphere

The three-dimensional FDTD simulation with perfectly matched layer (PML) boundary conditions was employed to calculate the electric field distribution of one CsPbBr₃ sphere at its emission wavelength. The refractive index of CsPbBr₃ was set as 2.4 according to the literature value.⁶⁶ Dipole source with its phase theta and phi of 0 was used as the light source, which was placed on the middle cross section of the CsPbBr₃ microsphere (viewed as X-Y plane). The simulation type of monitor was 2D-Z normal, and the monitored wavelength was set at 400–700 nm.

Conflicts of interest

There are no conflicts to declare.

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References

- X. Duan, Y. Huang, R. Agarwal and C. M. Lieber, *Nature*, 2003, **421**, 241–245.
- B. Guzelturk, Y. Kelestemur, K. Gungor, A. Yeltik, M. Z. Akgul, Y. Wang, R. Chen, C. Dang, H. Sun and H. V. Demir, *Adv. Mater.*, 2015, **27**, 2741.
- M. H. Huang, S. Mao, H. Feick, H. Yan, Y. Wu, H. Kind, E. Weber, R. Russo and P. Yang, *Science*, 2001, **292**, 1897–1899.
- A. L. Pan, D. Liu, R. B. Liu, F. F. Wang, X. Zhu and B. S. Zou, *Small*, 2005, **1**, 980–983.
- L. M. Tong, R. R. Gattass, J. B. Ashcom, S. L. He, J. Y. Lou, M. Y. Shen, I. Maxwell and E. Mazur, *Nature*, 2003, **426**, 816–819.
- Q. Zhang, X. Liu, M. I. B. Utama, G. Xing, T. C. Sum and Q. Xiong, *Adv. Mater.*, 2016, **28**, 276–283.
- A. Yadav, M. Danesh, L. Zhong, G. J. Cheng, L. Jiang and L. Chi, *Nanotechnology*, 2016, **27**, 165703.
- A. Yadav, L. Bai, Y. Yang, J. Liu, A. Kaushik, G. J. Cheng, L. Jiang, L. Chi and Z. Kang, *Nanoscale*, 2017, **9**, 5049–5054.
- Z. Junxi, Z. Lide and X. Wei, *J. Phys. D: Appl. Phys.*, 2012, **45**, 113001.
- I. Mora-Seró, F. Fabregat-Santiago, B. Denier, J. Bisquert, R. Tena-Zaera, J. Elias and C. Lévy-Clément, *Appl. Phys. Lett.*, 2006, **89**, 203117.
- P. D. Yang, H. Q. Yan, S. Mao, R. Russo, J. Johnson, R. Saykally, N. Morris, J. Pham, R. R. He and H. J. Choi, *Adv. Funct. Mater.*, 2002, **12**, 323–331.
- Q. Dong, Y. Fang, Y. Shao, P. Mulligan, J. Qiu, L. Cao and J. Huang, *Science*, 2015, **347**, 967–970.
- M. Liu, M. B. Johnston and H. J. Snaith, *Nature*, 2013, **501**, 395–398.
- D. Shi, V. Adinolfi, R. Comin, M. Yuan, E. Alarousu, A. Buin, Y. Chen, S. Hoogland, A. Rothenberger, K. Katsiev, Y. Losovyj, X. Zhang, P. A. Dowben, O. F. Mohammed, E. H. Sargent and O. M. Bakr, *Science*, 2015, **347**, 519–522.
- G. Xing, N. Mathews, S. Sun, S. S. Lim, Y. M. Lam, M. Gratzel, S. Mhaisalkar and T. C. Sum, *Science*, 2013, **342**, 344–347.
- S. De Wolf, J. Holovsky, S. J. Moon, P. Loper, B. Niesen, M. Ledinsky, F. J. Haug, J. H. Yum and C. Ballif, *J. Phys. Chem. Lett.*, 2014, **5**, 1035–1039.
- C. Shen, W. Du, Z. Wu, J. Xing, S. T. Ha, Q. Shang, W. Xu, Q. Xiong, X. Liu and Q. Zhang, *Nanoscale*, 2017, **9**, 8281.
- M. Gratzel, *Nat. Mater.*, 2014, **13**, 838–842.
- X. Liu, L. Niu, C. Wu, C. Cong, H. Wang, Q. Zeng, H. He, Q. Fu, W. Fu and T. Yu, *Adv. Sci.*, 2016, **3**, 1600137.
- Q. Zhang, R. Su, W. Du, X. Liu, L. Zhao, S. T. Ha and Q. Xiong, *Small Methods*, 2017, 1700163.
- Q. Zhang, R. Su, X. Liu, J. Xing, T. C. Sum and Q. Xiong, *Adv. Funct. Mater.*, 2016, **26**, 6238–6245.
- S. W. Eaton, M. Lai, N. A. Gibson, A. B. Wong, L. Dou, J. Ma, L. W. Wang, S. R. Leone and P. Yang, *Proc. Natl. Acad. Sci. U. S. A.*, 2016, **113**, 1993–1998.
- Y. Fu, H. Zhu, A. Schrader, D. Liang, Q. Ding, P. Joshi, L. Hwang, X. Y. Zhu and S. Jin, *Nano Lett.*, 2016, **16**, 1000.
- H. Zhu, Y. Fu, F. Meng, X. Wu, Z. Gong, Q. Ding, M. V. Gustafsson, M. T. Trinh, S. Jin and X. Zhu, *Nat. Mater.*, 2015, **14**, 636.
- J. Xing, X. F. Liu, Q. Zhang, S. T. Ha, Y. W. Yuan, C. Shen, T. C. Sum and Q. H. Xiong, *Nano Lett.*, 2015, **15**, 4571–4577.
- S. Zhang, Q. Shang, W. Du, J. Shi, Z. Wu, Y. Mi, J. Chen, F. Liu, Y. Li, M. Liu, Q. Zhang and X. Liu, *Adv. Opt. Mater.*, 2018, **6**, 1701032.
- W. Du, S. Zhang, J. Shi, J. Chen, Z. Wu, Y. Mi, Z. Liu, Y. Li, X. Sui, R. Wang, X. Qiu, T. Wu, Y. Xiao, Q. Zhang and X. Liu, *ACS Photonics*, 2018, **5**, 2051–2059.
- Y. Bekenstein, B. A. Koscher, S. W. Eaton, P. Yang and A. P. Alivisatos, *J. Am. Chem. Soc.*, 2015, **137**, 16008–16011.
- Y. Ling, Z. Yuan, Y. Tian, X. Wang, J. C. Wang, Y. Xin, K. Hanson, B. Ma and H. Gao, *Adv. Mater.*, 2016, **28**, 305–311.
- V. I. Klimov, A. A. Mikhailovsky, S. Xu, A. Malko, J. A. Hollingsworth, C. A. Leatherdale, H.-J. Eisler and M. G. Bawendi, *Science*, 2000, **290**, 314–317.
- J. Pan, S. P. Sarmah, B. Murali, I. Dursun, W. Peng, M. R. Parida, J. Liu, L. Sinatra, N. Alyami, C. Zhao, E. Alarousu, T. K. Ng, B. S. Ooi, O. M. Bakr and O. F. Mohammed, *J. Phys. Chem. Lett.*, 2015, **6**, 5027–5033.
- Y.-S. Park, W. K. Bae, T. Baker, J. Lim and V. I. Klimov, *Nano Lett.*, 2015, **15**, 7319–7328.
- S. Yakunin, L. Protesescu, F. Krieg, M. I. Bodnarchuk, G. Nedelcu, M. Humer, G. De Luca, M. Fiebig, W. Heiss and M. V. Kovalenko, *Nat. Commun.*, 2015, **6**, 8056.
- Q. Zhang, S. T. Ha, X. Liu, T. C. Sum and Q. Xiong, *Nano Lett.*, 2014, **14**, 5995–6001.
- D. Zhang, S. W. Eaton, Y. Yu, L. Dou and P. Yang, *J. Am. Chem. Soc.*, 2015, **137**, 9230–9233.
- K. Wu, G. Liang, Q. Shang, Y. Ren, D. Kong and T. Lian, *J. Am. Chem. Soc.*, 2015, **137**, 12792–12795.
- F. Deschler, M. Price, S. Pathak, L. E. Klintberg, D. D. Jarausch, R. Higler, S. Hüttner, T. Leijtens, S. D. Stranks, H. J. Snaith, M. Atature, R. T. Phillips and R. H. Friend, *J. Phys. Chem. Lett.*, 2014, **5**, 1421–1426.
- Z. Liu, J. Yang, J. Du, Z. Hu, T. Shi, Z. Zhang, Y. Liu, X. Tang, Y. Leng and R. Li, *ACS Nano*, 2018, **12**, 5923–5931.
- R. Agarwal, C. J. Barrelet and C. M. Lieber, *Nano Lett.*, 2005, **5**, 917.
- R. Matsuzaki, H. Soma, K. Fukuoka, K. Kodama, A. Asahara, T. Suemoto, Y. Adachi and T. Uchino, *Phys. Rev. B*, 2017, **96**, 125306.
- M. A. Versteegh, D. Vanmaekelbergh and J. I. Dijkhuis, *Phys. Rev. Lett.*, 2012, **108**, 157402.
- H. Haug and S. Koch, *Phys. Status Solidi B*, 2010, **82**, 531–543.
- C. Klingshirn and H. Haug, *Phys. Rep.*, 1981, **70**, 315–398.
- B. Tang, H. Dong, L. Sun, W. Zheng, Q. Wang, F. Sun, X. Jiang, A. Pan and L. Zhang, *ACS Nano*, 2017, **11**, 10681–10688.

- 45 B. Tang, L. X. Sun, W. H. Zheng, H. X. Dong, B. B. Zhao, Q. Q. Si, X. X. Wang, X. W. Jiang, A. L. Pan and L. Zhang, *Adv. Opt. Mater.*, 2018, **6**, 1800391.
- 46 X. Wang, H. Zhou, S. Yuan, W. Zheng, Y. Jiang, X. Zhuang, H. Liu, Q. Zhang, X. Zhu and X. Wang, *Nano Res.*, 2017, **10**, 3385–3395.
- 47 S. Hirotsu, J. Harada, M. Iizumi and K. Gesi, *J. Phys. Soc. Jpn.*, 1974, **37**, 1393–1398.
- 48 C. C. Stoumpos, C. D. Malliakas, J. A. Peters, Z. Liu, M. Sebastian, J. Im, T. C. Chasapis, A. C. Wibowo, D. Y. Chung, A. J. Freeman, B. W. Wessels and M. G. Kanatzidis, *Cryst. Growth Des.*, 2013, **13**, 2722–2727.
- 49 T. Schmidt, K. Lischka and W. Zulehner, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1992, **45**, 8989.
- 50 J. Butkus, P. Vashishtha, K. Chen, J. K. Gallaher, S. K. K. Prasad, D. Z. Metin, G. Laufersky, N. Gaston, J. E. Halpert and J. M. Hodgkiss, *Chem. Mater.*, 2017, **29**, 3644–3652.
- 51 S. Yakunin, L. Protesescu, F. Krieg, M. I. Bodnarchuk, G. Nedelcu, M. Humer, G. De Luca, M. Fiebig, W. Heiss and M. V. Kovalenko, *Nat. Commun.*, 2015, **6**, 8056.
- 52 K. Wang, W. Sun, J. Li, Z. Gu, S. Xiao and Q. Song, *ACS Photonics*, 2016, **3**, 1125–1130.
- 53 T. J. S. Evans, A. Schlaus, Y. Fu, X. Zhong, T. L. Atallah, M. S. Spencer, L. E. Brus, S. Jin and X. Y. Zhu, *Adv. Opt. Mater.*, 2018, **6**, 1700982.
- 54 K. Wei, Z. Xu, R. Chen, X. Zheng, X. Cheng and T. Jiang, *Opt. Lett.*, 2016, **41**, 3821–3824.
- 55 J. Lee, E. S. Koteles and M. O. Vassell, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1986, **33**, 5512–5516.
- 56 A. Ohtomo, K. Tamura, M. Kawasaki, T. Makino, Y. Segawa, Z. K. Tang, G. K. L. Wong, Y. Matsumoto and H. Koinuma, *Appl. Phys. Lett.*, 2000, **77**, 2204–2206.
- 57 L. Piskorski, R. P. Sarzała and W. Nakwaski, *Opto-Electron. Rev.*, 2011, **19**, 320–326.
- 58 O. Yaffe, Y. Guo, L. Z. Tan, D. A. Egger, T. Hull, C. C. Stoumpos, F. Zheng, T. F. Heinz, L. Kronik, M. G. Kanatzidis, J. S. Owen, A. M. Rappe, M. A. Pimenta and L. E. Brus, *Phys. Rev. Lett.*, 2017, **118**, 136001.
- 59 K. M. McCall, C. C. Stoumpos, S. S. Kostina, M. G. Kanatzidis and B. W. Wessels, *Chem. Mater.*, 2017, **29**, 4129–4145.
- 60 M. Balkanski, R. F. Wallis and E. Haro, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1983, **28**, 1928–1934.
- 61 P. K. Pandey, R. J. Choudhary, D. K. Mishra, V. G. Sathe and D. M. Phase, *Appl. Phys. Lett.*, 2013, **102**, 53–152.
- 62 P. Brüesch, *Anharmonicity*, Springer Berlin Heidelberg, 1982.
- 63 A. Bussmann-Holder and A. R. Bishop, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1991, **44**, 2853–2856.
- 64 C. Wolf and T.-W. Lee, *Mater. Today Energy*, 2018, **7**, 199–207.
- 65 M. S. Gaponenko, A. A. Lutich, N. A. Tolstik, A. A. Onushchenko, A. M. Malyarevich, E. P. Petrov and K. V. Yumashev, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **82**, 125320.
- 66 G. Murtaza and I. Ahmad, *Phys. B*, 2011, **406**, 3222–3229.