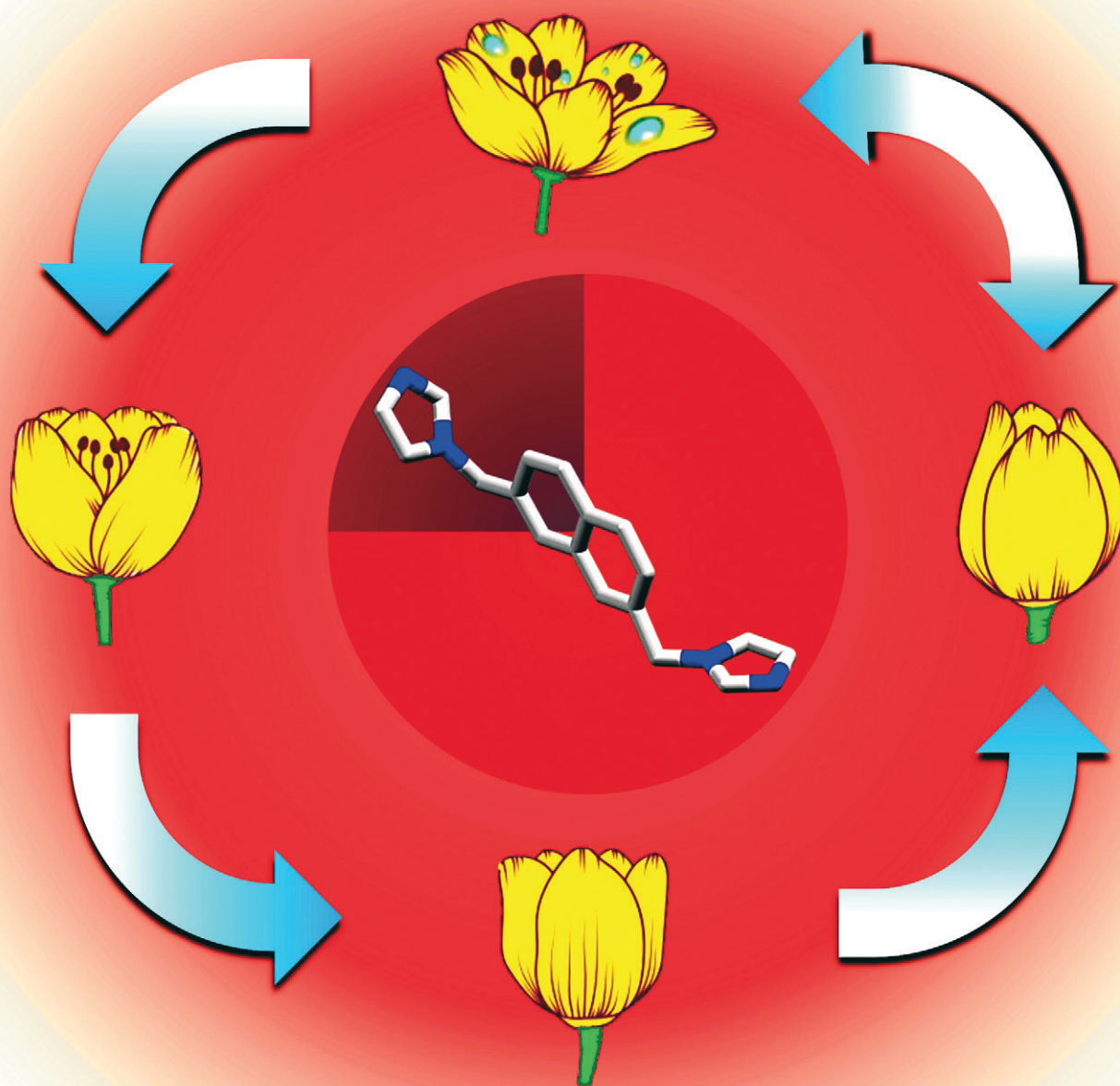


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PAPER

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Reversible phase transformations upon water uptake/removal in crystalline material of a bipodal N-donor ligand and evaluation of the stability of the hydrates formed†

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Isolation of the anhydrous form of 2,7-bis(imidazol-1-ylmethyl)-naphthalene (L, **1**) as well as three of its hydrates, in particular the hemihydrate (L·0.5H₂O, **2**), monohydrate (L·H₂O, **3**) and dihydrate (L·2H₂O, **4**), allowed to gain insight into the mechanism of the hydration process. SCXRD studies revealed that in the hemihydrate, the water molecules interact with only one N-donor site originating from the ligand and serve as a bridge between two ligand molecules, whereas in the monohydrate, the water molecules interact with both N-donor sites available in the bipodal ligand, leading to the formation of binary supramolecular 1D chains. In the dihydrate, the water molecules are forming 1D chains by themselves and simultaneously interact with the ligand to yield 2D supramolecular layers. The lattice energies as well as the stabilisation energies of the hydrates indicate a clear relationship between the stability of the structures and the hydration level. Supportive studies on bulk material, such as temperature dependent PXRD, as well as TGA were also performed. PXRD indicated a reversible transition between anhydrate formed upon heating and dihydrate formed after immersion of the sample in water. Water adsorption/desorption isotherms further confirmed that the crystalline material undergoes transformations induced by water molecules.

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1. Introduction

Hydrates are a class of multi-component crystalline solids, incorporating water into their crystal structure.¹ Their formation is largely driven by the presence of hydrogen bond donors and acceptors within an anhydrous compound. Desiraju reported that the formation of hydrates is particularly favourable when the host molecule has a low hydrogen bond donor-to-acceptor ratio, typically less than 0.5.² Later on, it was suggested that the presence of polar or charged functional groups in the molecule increases the likelihood of hydrate formation, as these groups enhance water interactions.³

Hydrates can be divided into three categories (i) channel hydrates, (ii) isolated site hydrates, and (iii) metal ion-

associated hydrates, based on structural features.^{4,5} Furthermore, there are two distinct types, namely stoichiometric and non-stoichiometric hydrates.⁶ However, this division can lead to confusion, as some authors include hydrates with a definite but non-integral ratio of water molecules in the group of non-stoichiometric hydrates⁷ and others restrict this subgroup to hydrates with no definite ratio of water molecules⁸ in relation to the host. The study of hydrates and their impact on the stability and structural diversity of molecular compounds has garnered substantial attention from both an experimental and theoretical point of view^{9–15} due to the pivotal roles that organic hydrates play in pharmaceutical formulations,^{16–19} and materials science.²⁰ Their significance in the pharmaceutical industry is connected with the fact that their occurrence can alter the physicochemical properties of a drug substance. However, while hydrate formation can be beneficial in pharmaceuticals, the presence of water poses substantial challenges in materials science, such as stability issues in metal-organic materials (MOMs) and decreased gas sorption performance in physisorbents.^{21,22}

Recently, computational and statistical tools have been employed to better understand hydrate formation^{23,24} and their properties, especially stability, complementing

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† Electronic supplementary information (ESI) available: Overlays of the ligands: 1/2, 1/3 and 1/4, PXRD patterns obtained for the products isolated from different solvents, thermogram of the bulk hydrated sample. CCDC reference numbers: 2400655–2400658. For ESI and crystallographic data in CIF or other electronic format see DOI: <https://doi.org/10.1039/d5ce00130g>

systematic studies on hydration and dehydration processes.^{25–28} Despite these advances, hydrate formation and their stability remain difficult to predict, posing an ongoing challenge in the fields mentioned above. Not without reason, Zaworotko and co-workers referred to hydrates as ‘the nemesis of crystal engineering’.²⁹

In here, we present a series of hydrates of a bipodal ligand containing a naphthalene core (Scheme 1), which were isolated by screening the crystallisation products from a range of solvents. Their structures were revealed by SCXRD. It is worth pointing out that based on a recent (2020) report, the number of hydrate–anhydrate pairs characterised *via* SCXRD studies is very limited. For example, only 167 pairs containing both the anhydrate and hemihydrate forms were deposited at CSD at that time.⁷ Recently (2024), the formation of seven hydrates was reported; however, five out of these show similar structural features and differ merely by the degree of hydration.³⁰ Furthermore, it was revealed that the hemi- and monohydrate which were isolated, are metastable phases, formed upon dehydration of the dihydrate. As the latter is a channel hydrate, such behaviour is rather uncommon, as usually such transitions take place between anhydrate and channel hydrate in one go and without significant structural rearrangements, resulting in similar unit cell parameters.³¹

2. Experimental

2.1. Reagents and materials

All commercially available chemicals were of reagent grade and were used without further purification. Anhydrous solvents were prepared using an INERT PureSolv solvent purification system.

2.2. Measurements

¹H and ¹³C NMR spectra were recorded in CDCl₃ on a Bruker Avance 700 MHz and 400 MHz instrument, respectively and referenced to residual solvent peaks. Melting points were collected with a Stuart SMP50 automated melting point apparatus, starting at a temperature of 30 °C and ramping up at 2 °C min^{−1}.

Thermal analysis (TGA, DTA) was performed on a TA Instruments SDT 650 analyser at a heating rate of 2 °C min^{−1} under dry nitrogen with a flow rate of 100 ml min^{−1} covering the temperature range: 25–600 °C.

PXRD patterns were obtained on a Philips X'Pert X-ray diffractometer using CuKα radiation. The samples were measured at the 2θ range of 4–45° with a scan speed of

4° min^{−1}. The PXRD data were analysed using the Powder Cell,³² Profex³³ and Match³⁴ software packages.

Isotherms were determined at a temperature of 25 °C using a gravimetric adsorption apparatus equipped with Baratron pressure transducers (MKS Instruments, Germany). Each sample was thermally desorbed before the measurement under high vacuum until a constant mass was obtained (after *ca.* 2 days).

2.3. Computational studies

2D fingerprint plots were generated with Crystal Explorer 17.0 by using a standard 0.6–2.4 Å range including reciprocal contacts.³⁵ The lattice energy calculations were performed at the (B3LYP/6-31G(d, p)) level of theory. The results were corrected for dispersion (Grimme D3 correction) and the basis set superposition error (BSSE). Ghost atoms used for the BSSE estimation were selected up to 5 Å distance from the considered molecule in the crystal lattice. The general methodology was analogous to that described in previous reports.³⁶ Before starting DFT calculations, O–H bonds positions were elongated to standard neutron mean values for all, determined by SCXRD structures of the hydrates.

2.4. Synthesis

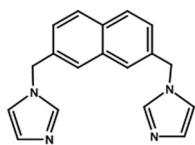
The synthesis of **L** was reported earlier as it was used during the preparation of naphthalenophane ligands.³⁷ However, we used an alternative procedure described below.

Imidazole (1.3 g, 19.08 mmol), dissolved in 20 ml dry THF, was slowly added to sodium hydride (60% in oil, 0.763 g, 19.08 mmol) in dry THF (20 mL) at 0 °C, and the resulting suspension was stirred at rt for 2 h (until the hydrogen gas evolution ceased). 2,7-Bis(bromomethyl)naphthalene (1 g, 3.18 mmol) solution in 25 ml dry THF was then slowly added and the resulting mixture was refluxed for another 12 h. The obtained mixture was dissolved in water and extracted with DCM (3 × 50 mL). The combined organic layers were washed with water (100 mL), dried over anhydrous magnesium sulphate and filtered. After removal of the solvent under reduced pressure, the product was re-dissolved in ethyl acetate and hexane was added dropwise until the solution became cloudy, and this was left at rt to precipitate the pure product as a yellowish solid (mass – 0.75 g, yield – 52%). Melting point – 138–141 °C.

¹H NMR (700 MHz, CDCl₃): δ 7.83–7.84 (2H, d), 7.61 (2H, s), 7.54 (2H, s), 7.28–7.30 (2H, d), 7.13 (2H, s), 6.95 (2H, s), 5.28 (4H, s).

¹³C NMR (100 MHz, CDCl₃): δ 137.52, 134.57, 133.19, 132.57, 129.97, 128.84, 126.16, 125.46, 119.37, 50.86.

The obtained pure compound was crystallized from solvents of different geometry and polarity, such as acetone, acetonitrile, DCM, MeOH, and THF, by dissolving 10 mg of the compound in 10 ml of each solvent. Vials covered with parafilm were left to undergo slow evaporation, which resulted in crystals of good quality for SCXRD in all vials.



Scheme 1 Representation of 2,7-bis(imidazol-1-ylmethyl)-naphthalene (**L**, 1).

Table 1 Crystal data and details of the refinement parameters for 1–4

Compound reference	1	2	3	4
Chemical formula	C ₁₈ H ₁₆ N ₄	C ₁₈ H ₁₇ N ₄ O _{0.5}	C ₁₈ H ₁₈ N ₄ O	C ₁₈ H ₂₀ N ₄ O ₂
Formula mass	288.35	297.35	306.36	324.38
Crystal system	Triclinic	Monoclinic	Orthorhombic	Monoclinic
<i>a</i> /Å	9.0492(3)	20.5410(4)	9.72620(10)	18.149(4)
<i>b</i> /Å	9.5052(4)	5.46830(10)	10.12810(10)	9.6466(19)
<i>c</i> /Å	9.9724(3)	26.6872(5)	31.3817(5)	9.3799(19)
α /°	62.525(4)	90	90	90
β /°	75.236(3)	102.478(2)	90	93.06(3)
γ /°	80.607(3)	90	90	90
Unit cell volume/Å ³	734.86(5)	2926.82(10)	3091.35(7)	1639.9(6)
Temperature/K	100(2)	100(2)	100(2)	100(2)
Space group	<i>P</i> $\bar{1}$	<i>C</i> 2/ <i>c</i>	<i>Pbca</i>	<i>P</i> 2 ₁ / <i>c</i>
No. of formula units per unit cell, <i>Z</i>	2	8	8	4
Radiation type	CuK α	CuK α	CuK α	CuK α
Absorption coefficient, μ /mm ^{−1}	0.632	0.676	0.680	0.089
No. of reflections measured	26 217	10 042	26 217	8358
No. of independent reflections	3044	2997	3187	2892
<i>R</i> _{int}	0.0441	0.0307	0.0644	0.0470
Final <i>R</i> ₁ ^{<i>a</i>} values (<i>I</i> > 2 σ (<i>I</i>))	0.0391	0.0371	0.0383	0.0581
Final <i>wR</i> ₂ ^{<i>b</i>} values (<i>I</i> > 2 σ (<i>I</i>)) ^{<i>b</i>}	0.1008	0.0936	0.1030	0.1174
Final <i>R</i> ₁ ^{<i>a</i>} values (all data)	0.0444	0.0421	0.0428	0.0880
Final <i>wR</i> ₂ ^{<i>b</i>} values (all data)	0.1042	0.0973	0.1061	0.1293
Goodness of fit on <i>F</i> ²	1.056	1.048	1.039	1.057

$$^a R_1 = \sum ||F_o| - |F_c|| / \sum |F_o|. \quad ^b wR_2 = \{ \sum [w(F_o^2 - F_c^2)^2] / \sum [w(F_o^2)^2] \}^{1/2}.$$

2.5. Structure determination

X-ray crystallographic data for 1–4 were collected on a Rigaku XtaLab Synergy-S diffractometer equipped with graphite-monochromated CuK α radiation (λ = 1.5406 Å). The crystals were mounted on a loop upon coating with paratone-N oil. Data collection was carried out at 100(2) K to minimize solvent loss, possible structural disorder and thermal motion effects. Data frames were processed (unit cell determination, intensity data integration, correction for Lorentz and polarisation effects, and empirical absorption correction) by using the corresponding diffractometer's software package, CrysAlisPro.³⁸ The structures were solved by using direct methods with SHELXS-2019/3 (ref. 39) and refined by using full-matrix least-squares methods based on *F*² by using SHELXL-2019/3 (ref. 40) using the Olex⁴¹ interface. The programs Mercury⁴² and POV-Ray⁴³ were both used to prepare molecular graphics. All non-hydrogen atoms were refined anisotropically. All hydrogen atoms were positioned geometrically with C–H = 0.95 Å (aromatic) and 0.99 Å (methylene) and refined as riding, with *U*_{iso}(H) = 1.2*U*_{eq}(C), except the water hydrogen atoms which were located in a difference map and refined freely in case of 2 and 3 and with restrained O–H bond lengths in case of 4. A summary of the data collection and structure refinement parameters is provided in Table 1.

3. Results and discussion

Crystallization of **L** from a range of solvents allowed us to isolate a series of hydrates: hemi- (2), mono- (3) and dihydrate (4). Interestingly, none any polymorph of the

ligand was isolated during these screenings, which was a case for the related ligands containing more flexible biphenyl core.⁴⁴

3.1. Crystal structures

1 crystallizes in the centrosymmetric space group of the triclinic system, with one molecule in the asymmetric unit. Its molecular structure is shown in Fig. 1. The angle between the mean planes of the imidazole rings is 52.6° and the angles between the mean planes of the imidazole rings and the naphthalene core are 84.13° and 89.04°, respectively. The molecules form rows extending along the *a* axis in which they adopt orientations alternating by an inversion centre. There are numerous C–H⋯N hydrogen bonds, resulting in the formation of a 3D supramolecular assembly, which is further stabilized by C–H⋯ π interactions involving the methylene groups, naphthalene core and imidazole ring as donors and the naphthalene core, and imidazole rings as acceptors of these hydrogen bonds (Table 2).

2 crystallizes as a hemihydrate in the *C*2/*c* space group of the monoclinic system, wherein the asymmetric unit contains one ligand molecule and half a water molecule, as the O atom is located on an inversion centre (Fig. 2). The angle



Fig. 1 Molecular structure of **1** with atomic displacement plot shown at 50% probability.

Table 2 Hydrogen bonding parameters for 1–4

Compound	D–H···A	H···A (Å)	D···A (Å)	D–H–A (°)	Compound	D–H···A	H···A (Å)	D···A (Å)	D–H–A (°)
1	C8–H8···N3 ⁱ	2.61	3.532(2)	164	3	O23–H23B···N3 ⁱ	2.01	2.887(1)	170
	C13–H13···N3 ⁱⁱ	2.76	3.708(1)	172		O23–H23A···N20 ⁱⁱ	1.99	2.873(1)	178
	C21–H21···N3 ⁱⁱⁱ	2.77	3.605(2)	148		C6–H6A···N3 ⁱⁱⁱ	2.67	3.649(1)	170
	C22–H22···N3 ^{iv}	2.99	3.915(2)	165		C9–H9···N3 ^{iv}	2.88	3.741(1)	152
	C4–H4···N20 ^v	2.90	3.675(2)	139		C16–H16···N20 ^v	2.90	3.538(1)	125
	C5–H5···N20 ^{vi}	2.69	3.557(2)	156		C17–H17A···N20 ^{vi}	2.75	3.503(2)	133
	C12–H12···N20 ^{vii}	2.53	3.471(1)	169		C4–H4···O23 ^{vii}	2.55	3.371(2)	145
	C17–H17B···N20 ^{viii}	2.76	3.446(1)	127		C9–H9···O23 ^{iv}	2.85	3.666(1)	145
	C13–H13···Cg1 ⁱⁱ	2.76	3.599(1)	148		C13–H13···O23 ^{iv}	2.52	3.421(1)	158
	C6–H6A···Cg3 ^{iv}	2.62	3.524(2)	151		C16–H16···O23 ⁱⁱⁱ	2.92	3.802(1)	155
	C19–H19···Cg3 ^{vi}	2.59	3.517(1)	166		C21–H21···O23 ^{viii}	2.33	3.143(2)	143
	C2–H2···Cg4 ^{iv}	2.89	3.780(1)	156	4	O23–H23A···N3 ⁱ	1.93	2.856(3)	165
	O23–H23A···N3 ⁱ	1.99	2.896(1)	156		O23–H23B···O24 ⁱ	1.82	2.754(3)	174
	O23–H23B···N3 ⁱⁱ	1.94	2.896(1)	168		O24–H24B···N20 ⁱⁱ	1.82	2.776(2)	168
	C17–H17A···N3 ⁱⁱⁱ	2.76	3.723(2)	164		O24–H24A···O23 ⁱⁱⁱ	1.86	2.791(3)	167
	C2–H2···N18 ^{iv}	2.67	3.399(2)	134		C17–H17A···N20 ^{iv}	2.71	3.693(4)	170
	C9–H9···N20 ^v	2.82	3.632(2)	144		C4–H4···O23 ⁱⁱⁱ	2.93	3.784(3)	150
	C21–H21···N20 ^{vi}	2.81	3.571(1)	138		C6–H6A···O23 ^v	2.71	3.515(3)	138
	C22–H22···N20 ^{vii}	2.64	3.439(2)	142		C12–H12···O23 ^v	2.91	3.712(3)	143
	C16–H16···O23 ⁱⁱⁱ	2.94	3.787(1)	149		C16–H16···O23 ^{vi}	2.85	3.412(3)	118
	C17–H17B···O23 ^{viii}	2.53	3.273(1)	132		C9–H9···O24 ^{vii}	2.72	3.585(3)	153
	C2–H2···Cg2 ^{iv}	2.71	3.372(1)	128		C12–H12···O24 ^v	2.81	3.686(3)	153
	C19–H19···Cg4 ^{ix}	2.85	3.753(1)	160		C13–H13···O24 ^{vii}	2.77	3.624(3)	150
2	O23–H23A···N3 ⁱ	1.99	2.896(1)	156		C16–H16···O24 ^v	2.82	3.692(3)	153
	O23–H23B···N3 ⁱⁱ	1.94	2.896(1)	168		C21–H21···O24 ^{viii}	2.74	3.548(3)	144
	C17–H17A···N3 ⁱⁱⁱ	2.76	3.723(2)	164					
	C2–H2···N18 ^{iv}	2.67	3.399(2)	134					
	C9–H9···N20 ^v	2.82	3.632(2)	144					
	C21–H21···N20 ^{vi}	2.81	3.571(1)	138					
	C22–H22···N20 ^{vii}	2.64	3.439(2)	142					
	C16–H16···O23 ⁱⁱⁱ	2.94	3.787(1)	149					
	C17–H17B···O23 ^{viii}	2.53	3.273(1)	132					
	C2–H2···Cg2 ^{iv}	2.71	3.372(1)	128					
	C19–H19···Cg4 ^{ix}	2.85	3.753(1)	160					

(1) Cg1 is the centroid of the imidazole ring N1–C5, Cg3 is the centroid of the benzene ring C7–C12, Cg4 is the centroid of the benzene ring C10–C16, (2) Cg2 is the centroid of the imidazole ring N18–C22, Cg4 is the centroid of the benzene ring C10–C16, Symmetry codes (1): (i) $1 - x, 2 - y, 1 - z$, (ii) $x, y, z - 1$, (iii) $x, y - 1, z - 1$, (iv) $1 - x, 1 - y, 1 - z$, (v) $x, 1 + y, 1 + z$, (vi) $2 - x, 1 - y, -z$, (vii) $x, y, 1 + z$, (viii) $2 - x, -y, -z$; (2): (i) x, y, z , (ii) $-x, y, 1/2 - z$, (iii) $1/2 + x, 1/2 + y, z$, (iv) $x - 1/2, y - 3/2, z$, (v) $1 - x, 2 - y, 1 - z$, (vi) $3/2 - x, 5/2 - y, 1 - z$, (vii) $x, y - 1, z$, (viii) $1/2 + x, 3/2 + y, z$, (ix) $x, 1 + y, z$; (3): (i) x, y, z , (ii) $3/2 - x, 1 - y, z - 1/2$, (iii) $x - 1/2, 3/2 - y, 1 - z$, (iv) $1/2 + x, 3/2 - y, 1 - z$, (v) $1 - x, 1/2 + y, 3/2 - z$, (vi) $x - 1/2, y, 3/2 - z$, (vii) $3/2 - x, y - 1/2, z$, (viii) $x, 3/2 - y, 1/2 + z$; (4): (i) x, y, z , (ii) $x - 1, y, z - 1$, (iii) $x, 3/2 - y, z - 1/2$, (iv) $2 - x, 1/2 + y, 3/2 - z$, (v) $1 - x, 1/2 + y, 1/2 - z$, (vi) $1 - x, 2 - y, 1 - z$, (vii) $1 - x, y - 1/2, 1/2 - z$, (viii) $1 + x, 3/2 - y, 1/2 + z$.

between the mean planes of the imidazole rings is 46.42° and the angles between the mean planes of the imidazole rings and the naphthalene core are 77.35° and 84.71° , respectively, revealing certain conformational differences in comparison with the anhydrate form (Fig. S1†). A water molecule is interconnecting two ligands *via* strong O–H···N hydrogen bonds, namely O23–H23B···N3 with an O···N distance of $2.897(1)$ Å and an O–H–N angle of 170° . Furthermore, it also acts as a hydrogen bond acceptor, participating in C–H···O interactions and yielding 2D supramolecular layers expanding along the *a*–*b* plane, which are further stabilised *via* C–H··· π interactions. These are further extended in a third direction *via* weak C–H···N hydrogen bonds (see Table 2).

3 crystallizes as a monohydrate in the *Pbca* space group of the orthorhombic system with one ligand molecule and one water molecule in the asymmetric unit (Fig. 3).

The angle between the mean planes of the imidazole rings is 54.20° and the angles between the mean planes of the imidazole rings and the naphthalene core are 89.89° and 86.02° , respectively, indicating more similarity with 1 than with 2 (Fig. S1†).

In this case, the water molecule interacts with both imidazole rings of the ligand molecule *via* strong O–H···N hydrogen bonds, O23–H23B···N3 (O···N distance of $2.887(1)$ Å, O–H–N angle of 170°) and O23–H23A···N20 (O···N distance of $2.873(1)$ Å, O–H–N angle of 178°), forming 1D zig-zag chains, which are extended into a 3D assembly *via*

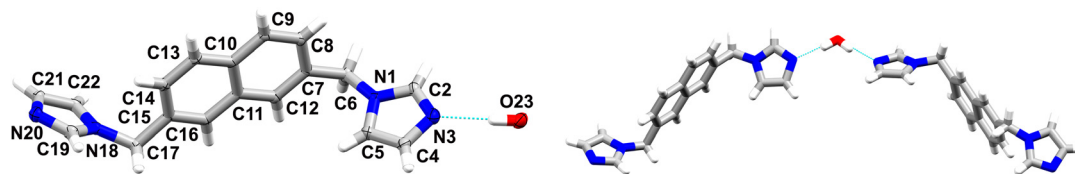


Fig. 2 On the left: representation of the asymmetric unit of 2 with atomic displacement plot shown at 50% probability; on the right: the supramolecular discrete motive stabilised by a water molecule.

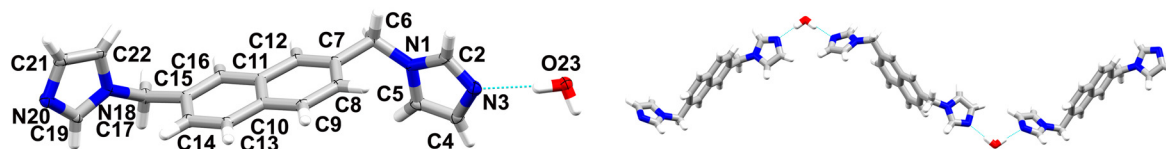


Fig. 3 On the left: molecular structure of **3** with atomic displacement plot shown at 50% probability; on the right: fragment of the 1D supramolecular chain, bridged by water molecules.

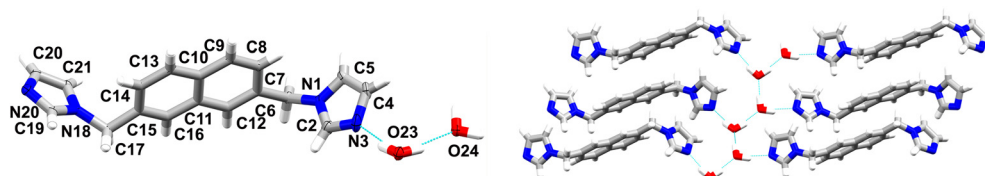


Fig. 4 On the left: molecular structure of **4** with atomic displacement plot shown at 50% probability; on the right: fragment of the 2D supramolecular layers, bridged by water molecules.

C–H $\cdots$ O hydrogen bonds. This assembly is further stabilised through C–H $\cdots$ N hydrogen bonds and π – π interactions, which involve adjacent naphthalene cores with a centroid–centroid distance of *ca.* 3.8 Å. Interestingly, C–H $\cdots$ π interactions are absent in this hydrate.

4 crystallizes as a dihydrate in the $P2_1/c$ space group of the monoclinic system. It shows the presence of one ligand molecule and two water molecules in the asymmetric unit (Fig. 4). The angle between the mean planes of the imidazole rings is 63.73° and the angles between the mean planes of the imidazole rings and the naphthalene core are 83.85° and 87.01°, respectively. The water molecules interact with each other in this case, forming 1D chiral chains along the *c* axis *via* O23–H23B $\cdots$ O24 (O $\cdots$ O distance of 2.754(3) Å, O–H–O angle of 174°), and O24–H24A $\cdots$ O23 (O $\cdots$ O distance of 2.791(3) Å, O–H–O angle of 167°) hydrogen bonding. These chains further interact with the ligand molecules *via* O–H $\cdots$ N hydrogen bonds, with O–N distances of *ca.* 2.8 Å and O–H–N angles oscillating around 167°, yielding 2D layers in the *a*–*c* plane which are stabilised *via* π – π interactions between adjacent naphthalene cores with a centroid–centroid distance of *ca.* 3.7 Å. These layers are extended in the third dimension *via* weak C–H $\cdots$ O hydrogen bonds and further stabilised *via* C–H $\cdots$ N hydrogen bonds.

Like in **3**, C–H $\cdots$ π interactions are absent. Interestingly, it was noticed before during studies on hydrates of five and six-membered N-heterocyclic aromatic compounds containing hydrogen bond acceptors, that hydrates in which water formed 1D chains showed a preference to form π – π interactions.¹⁰ The relative contributions (percentages) of selected intermolecular contacts to the Hirshfeld surface were calculated to further analyse the distribution of the intermolecular interactions. As shown in Fig. 5, the C–H $\cdots$ N and C–H $\cdots$ π hydrogen bonds gradually disappear upon hydration in favour of hydrogen bonds involving the O atoms and π – π interactions (see also Table 2).

3.2. Study on bulk material and hydrate conversions

PXRD studies indicated that all samples, isolated from different solvents, are mixtures containing products of different degrees of hydration (Fig. S2†), the relative contributions of which are influenced by time and storage conditions. However, upon heating the samples for a couple of minutes at 100 °C, the bulk material transforms to the anhydrate, which can be hydrated again by immersion of a powdered sample in water (Fig. 6).

TGA of the hydrated samples indicates that dehydration occurs in separate steps: the first up to approximately 45 °C, the next up to 70 °C, and the last up to around 141 °C (melting), showcasing the gradual process of dehydration (Fig. S3†).

This observation was further confirmed by collecting the adsorption/desorption isotherms of H₂O. The adsorption branch follows an almost horizontal course up to a relative pressure of about 0.8 *p/p*₀, which could indicate that the tested material is rather hydrophobic in nature. When the relative pressure of water vapor exceeds the value of 0.85, the structure reorganizes, allowing capillary condensation and reaching near-maximum adsorption. The observed type V adsorption isotherm results from the formation of only a

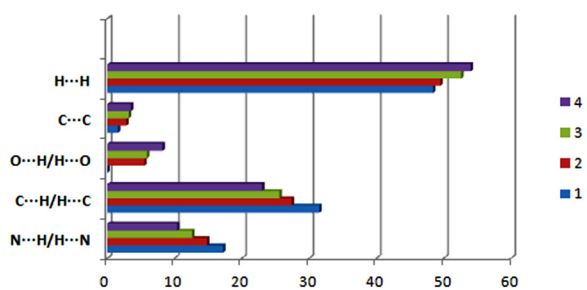


Fig. 5 The relative contributions (percentages) of selected intermolecular contacts to the Hirshfeld surface.

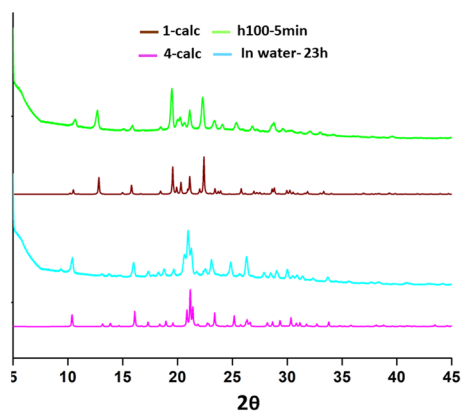


Fig. 6 Exemplary experimental PXRD patterns of a sample obtained by crystallisation from THF: heated for 5 minutes at 100 °C, immersed for 23 h in water upon heating, and the simulated PXRD patterns for the anhydrate **1** and the dihydrate **4**, respectively.

limited number of adsorption layers within the adsorbent pores due to their hydrophobicity. It is evident that the adsorbent–adsorbate interactions are significantly weaker than the adsorbate–adsorbate interactions, leading to condensation in the adsorbent pores at high relative pressures, resulting in significant hysteresis. The formation of bridged water molecules *via* hydrogen bonds in micropores is the main reason for the observed phenomenon. Interestingly, the desorption branch of the isotherm clearly reflects the gradual nature of the process, specifically the transitions between forms with different degrees of hydration (Fig. 7). The plateau observed at maximum relative pressure corresponds to the presence of *ca.* 1.7 adsorbate molecules in the adsorbent structure (the discrepancy between the number of water molecules in the crystal structure obtained from the SCXRD and the adsorption studies is most likely caused by the statistical distribution of crystallite grains). The next plateau appears at a relative pressure of about 0.7 p/p_0 and corresponds to the presence of the monohydrate (about 1.1 molecules of H₂O per ligand). At a relative pressure below 0.15 p/p_0 , partial desorption is observed, leading to the third plateau whereby approximately 0.5 molecules of H₂O are associated

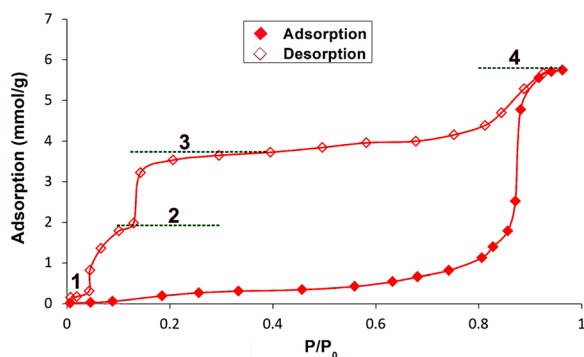


Fig. 7 Water adsorption/desorption isotherms as a function of relative pressure, with markings indicating the formation of **1–4**.

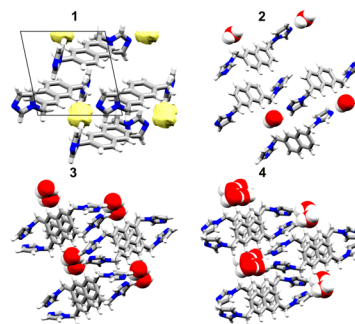


Fig. 8 Packing diagrams of **1–4** shown along the *b* (**1–3**) and *c* (**4**) axis, small apertures present in **1** indicated in yellow, water molecules shown in space filling representation.

with each molecule of the adsorbent (hemihydrate formation).

The revealed structural transitions taking place in the bulk let us have one more look at the crystal structures **1–4**, whereby crystal packing similarities, despite changes in crystallographic parameters (Table 1), support the gradual mechanism of the dehydration process. The process is connected with a significant reorganization on molecular level by forming new intermolecular interactions upon water removal, achieved through molecular shifts and small conformational adjustments of the ligand (Fig. 8), the changes of which were followed in a single-crystal to single-crystal fashion.

There are small apertures in the anhydrous form (that could be mapped with a probe of 0.9 Å per contact surface), which could be sufficient to initiate water uptake, as a related phenomenon was observed in the case of an apohost of an organic macrocycle undergoing reversible uptake of THF in a single-crystal.⁴⁵ However, immersion of anhydrous bulk material in water leads to the formation of the dihydrate form *via* recrystallization, as shown in Fig. S4.†

3.3. Crystal lattice energy calculations and stabilisation energy of hydrates

Hydrates exhibit different thermodynamic stability and solubility compared to their pure anhydrous forms. Typically, the introduction of water into the crystal lattice leads to a decrease in lattice energy per unit cell, as water molecules can form additional intermolecular hydrogen bonds, increasing the number of interactions. This can also be observed for the hydrates presented here, for which the lattice energy decreases upon hydration (Table 3). Furthermore, the stabilisation energy of the hydrates (E_{stab})

Table 3 The results of crystal lattice energy calculations and the stabilisation energy of the hydrates

	1	2	3	4
$E_{\text{latt}}/\text{per unit cell (kJ mol}^{-1}\text{)}$	−198.7	−263.3	−266.9	−365.8
$E_{\text{stab}} \text{ (kJ mol}^{-1}\text{)}$		−67.0	−68.2	−83.6

was determined following the procedure reported by Braun *et al.*⁴⁶ whereby E_{stab} was calculated in order to predict the occurrence of particular hydrates. The stabilisation energy relates the different hydrates stoichiometries to the anhydrate and ice. By evaluating this energy, it is possible to determine which hydrates are more thermodynamically stable compared to the anhydrous form. The lattice energy of ice was calculated at the same theory level as in mentioned above report, varying in our case from -60 to -64 kJ mol⁻¹ for the ordered ice polymorphs.⁴⁷ All determined E_{stab} (Table 3) are lower than the lattice energies calculated for the ice polymorphs, suggesting that the hydrates obtained are thermodynamically stable, which is in agreement with the experimental results.

Conclusions

The crystalline material based on bipodal heterocyclic ligand with rigid naphthalene core appears to undergo a series of transformations induced by water uptake/removal. The process is connected with the molecular reorganisation, whereby weak C-H...N and C-H... π hydrogen bonds present in the anhydrate are replaced with strong O-H...N and O-H...O hydrogen bonds upon water incorporation into the crystalline material and dihydrate formation (channel hydrate; formed in water *via* recrystallization of the anhydrous form). Upon desorption, this channel hydrate gradually transforms to the anhydrate *via* mono- and hemihydrate phases, which is clearly reflected in the desorption branch of the H₂O isotherm and it is possible to follow these transformations in a single-crystal. The crystal lattice energies show significant differences between the most stable dihydrate and the metastable phases of mono- and hemihydrate, which could be the reason that a one-step adsorption isotherm is observed, whereas the stabilisation energy confirms the feasibility of isolating all three hydrates experimentally. Computational studies, allowing to model such transitions, would provide valuable insights, but they are not yet feasible.

Data availability

The data supporting this article have been included as part of the ESI.†

Conflicts of interest

There are no conflicts to declare.

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References

- 1 C. H. Görbitz and H. P. Hersleth, *Acta Crystallogr., Sect. B: Struct. Sci.*, 2000, **56**, 526–534.
- 2 G. R. Desiraju, *J. Chem. Soc., Chem. Commun.*, 1991, 426–428.
- 3 L. Infantes, L. Fábián and W. D. S. Motherwell, *CrystEngComm*, 2007, **9**, 65–71.
- 4 K. R. Morris, *Polymorphism in Pharmaceutical Solids*, ed. H. G. Brittain, Marcel Dekker, Inc., New York, 1999, pp. 125–181.
- 5 K. R. Morris and N. Rodriguez-Hornedo, *Encyclopedia of Pharmaceutical Technology*, ed. J. Swarbrick and J. C. Boylan, Dekker, New York, 1993, vol. 7, pp. 393–441.
- 6 U. J. Griesser, *Polymorphism in the Pharmaceutical Industry*, ed. R. Hilfiker, Wiley-VCH, Weinheim, 2006, pp. 211–257.
- 7 J. E. Werner and J. A. Swift, *CrystEngComm*, 2020, **22**, 7290–7297.
- 8 E. Tieger, V. Kiss, G. Pokol, Z. Finta, M. Dušek, J. Rohlíček, E. Skořepová and P. Brázda, *CrystEngComm*, 2016, **18**, 3819–3831.
- 9 A. Bajpai, H. S. Scott, T. Pham, K. J. Chen, B. Space, M. Lusi, M. L. Perry and M. J. Zaworotko, *IUCrJ*, 2016, **3**, 430–439.
- 10 R. Sanii, E. Patyk-Kaźmierczak, C. Hua, S. Darwish, T. Pham, K. A. Forrest, B. Space and M. J. Zaworotko, *Cryst. Growth Des.*, 2021, **21**, 4927–4939.
- 11 S. K. Rai, A. Gunnam, G. J. O. Beran, J. A. Kaduk and A. K. Nangia, *Cryst. Growth Des.*, 2023, **23**, 1179–1188.
- 12 P. Klitou, I. Rosbottom and E. Simone, *Cryst. Growth Des.*, 2019, **19**, 4774–4783.
- 13 A. T. Hulme and S. L. Price, *J. Chem. Theory Comput.*, 2007, **3**, 1597–1608.
- 14 R. Mizoguchi and H. Uekusa, *Cryst. Growth Des.*, 2018, **18**, 6142–6149.
- 15 A. A. Malik and A. A. Dar, *J. Mol. Struct.*, 2024, **1310**, 138254.
- 16 E. Jurczak, A. H. Mazurek, Ł. Szeleszczuk, D. M. Pisklak and M. Zielińska-Pisklak, *Pharmaceutics*, 2020, **12**, 1–25.
- 17 P. Chakravarty and K. Nagapudi, *TrAC, Trends Anal. Chem.*, 2021, **140**, 116276.
- 18 A. M. Healy, Z. A. Worku, D. Kumar and A. M. Madi, *Adv. Drug Delivery Rev.*, 2017, **117**, 25–46.
- 19 R. K. Khankari and D. J. W. Grant, *Thermochim. Acta*, 1995, **248**, 61–79.
- 20 *Clathrate Hydrates: Molecular Science and Characterization*, ed. J. A. Ripmeester and S. Alavi, John Wiley & Sons, 2022.
- 21 A. Kumar, D. G. Madden, M. Lusi, K. J. Chen, E. A. Daniels, T. Curtin, J. J. Perry and M. J. Zaworotko, *Angew. Chem., Int. Ed.*, 2015, **54**, 14372–14377.
- 22 Y. Ming, J. Purewal, J. Yang, C. Xu, R. Soltis, J. Warner, M. Veenstra, M. Gaab, U. Müller and D. J. Siegel, *Langmuir*, 2015, **31**, 4988–4995.
- 23 S. Boothroyd, A. Kerridge, A. Broo, D. Buttar and J. Anwar, *Cryst. Growth Des.*, 2018, **18**, 1903–1908.
- 24 D. E. Braun and U. J. Griesser, *Cryst. Growth Des.*, 2016, **16**, 6405–6418.
- 25 R. Bobrovs, A. Kons, A. Berziņš, T. Rekis and A. Actiņš, *Cryst. Growth Des.*, 2018, **18**, 2100–2111.

- 26 R. Okura, H. Uchiyama, K. Kadota and Y. Tozuka, *CrystEngComm*, 2021, **23**, 4816–4824.
- 27 D. E. Braun, J. A. McMahon, R. M. Bhardwaj, J. Nyman, M. A. Neumann, J. Van De Streek and S. M. Reutzel-Edens, *Cryst. Growth Des.*, 2019, **19**, 2947–2962.
- 28 D. E. Braun, L. H. Koztecki, J. A. McMahon, S. L. Price and S. M. Reutzel-Edens, *Mol. Pharmaceutics*, 2015, **12**, 3069–3088.
- 29 H. D. Clarke, K. K. Arora, H. Bass, P. Kavuru, T. T. Ong, T. Pujari, L. Wojtas and M. J. Zaworotko, *Cryst. Growth Des.*, 2010, **10**, 2152–2167.
- 30 O. Kaszubowski, K. Ślepokura, K. N. Truong and J. Wojciechowski, *Cryst. Growth Des.*, 2024, **24**, 8516–8527.
- 31 S. Kuang, H. S. Shah and B. Zhao, *Pharm. Res.*, 2024, **41**, 1533–1541.
- 32 W. Kraus and G. Nolze, *J. Appl. Crystallogr.*, 1996, **29**, 301–303.
- 33 N. Doebelin and R. Kleeberg, *J. Appl. Crystallogr.*, 2015, **48**, 1573–1580.
- 34 Match! - Phase Analysis using Powder Diffraction, Crystal Impact - Dr. H. Putz & Dr. K. Brandenburg GbR, Kreuzherrenstr. 102, 53227 Bonn, Germany, <https://www.crystalimpact.de/match>.
- 35 P. R. Spackman, M. J. Turner, J. J. McKinnon, S. K. Wolff, D. J. Grimwood, D. Jayatilaka and M. A. Spackman, *J. Appl. Crystallogr.*, 2021, **54**, 1006–1011.
- 36 S. Feng and T. Li, *J. Chem. Theory Comput.*, 2006, **2**, 149–156.
- 37 C. Caron, X. N. T. Duong, R. Guillot, S. Bombard and A. Granzhan, *Chem. – Eur. J.*, 2019, **25**, 1949–1962.
- 38 Rigaku Oxford Diffraction, *CrysAlisPro Software System, version 1.171.38.41*, Rigaku Corporation, Oxford, UK, 2015.
- 39 G. M. Sheldrick, *Acta Crystallogr., Sect. A: Found. Adv.*, 2015, **71**, 3–8.
- 40 G. M. Sheldrick, *Acta Crystallogr., Sect. C: Struct. Chem.*, 2015, **71**, 3–8.
- 41 O. V. Dolomanov, L. J. Bourhis, R. J. Gildea, J. A. K. Howard and H. Puschmann, *J. Appl. Crystallogr.*, 2009, **42**, 339–341.
- 42 C. F. Macrae, I. J. Bruno, J. A. Chisholm, P. R. Edgington, P. McCabe, E. Pidcock, L. Rodriguez-Monge, R. Taylor, J. van de Streek and P. A. Wood, *J. Appl. Crystallogr.*, 2008, **41**, 466–470.
- 43 <http://www.povray.org/>.
- 44 S. Chaudhary, D. Kędziera, Z. Rafiński and L. Dobrzańska, *RSC Adv.*, 2023, **13**, 30625–30632.
- 45 J. Thomas, G. Reekmans, P. Adriaenssens, L. Van Meervelt, M. Smet, W. Maes, W. Dehaen and L. Dobrzańska, *Angew. Chem., Int. Ed.*, 2013, **52**, 10237–10240.
- 46 D. E. Braun, P. G. Karamertzanis and S. L. Price, *Chem. Commun.*, 2011, **47**, 5443–5445.
- 47 F. Della Pia, A. Zen, D. Alfè and A. Michaelides, *J. Chem. Phys.*, 2022, **157**, 134701.