

CrossMark
click for updatesCite this: *RSC Adv.*, 2017, 7, 415

High-performance all-solid-state flexible carbon/TiO₂ micro-supercapacitors with photo-rechargeable capability†

Jinguang Cai,^{ab} Chao Lv^{ab} and Akira Watanabe^{*a}

A high-performance all-solid-state flexible interdigitated carbon/TiO₂ micro-supercapacitor (MSC) with photo-rechargeable capability was prepared by combining a laser direct writing technique with electrophoretic deposition of TiO₂ nanoparticles. The carbon/TiO₂ MSC shows the same excellent capacitive performance as the pure carbon MSC, such as high specific capacitance up to 27.3 mF cm⁻² at a typical current density of 0.05 mA cm⁻², excellent cycling stability, long-time stability, and mechanical stability. Under UV light irradiation, the carbon/TiO₂ MSC can be charged to above 100 mV, and still maintain 60 mV after 10 photo-charging cycles, demonstrating its photo-rechargeable capability. What's more, after 10 photo-charging cycles, the carbon/TiO₂ MSC shows no capacitive performance degradation, which can be attributed to the high structural stability of the carbon/TiO₂ MSC under UV light irradiation.

Received 12th October 2016
Accepted 4th November 2016

DOI: 10.1039/c6ra25136f

www.rsc.org/advances

1. Introduction

Recently, miniaturized portable and wearable flexible electronic devices have aroused worldwide research attention, but the relatively high volume and weight of the energy supply units limit the whole size of the micro-system.^{1,2} Therefore, it is important to develop micro-energy storage units with small sizes and high energy and power densities, as well as integrated energy harvesting units, to make the whole micro-system self-powered. Micro-supercapacitors (MSCs) are recognized as potential power supply units for on-chip micro-devices because they possess not only the advantages of supercapacitors such as high power density, excellent cycling performance, pollution-free operation, maintenance-free features, small size, light weight, and flexibility, but also simplified packaging processes, robust all solid state, and compatibility with integrated circuits.²⁻⁷ Commercial solar panels can be used to charge the MSCs, making the system self-powered, but commercial solar panels usually have heavy weight and large size, limiting the miniaturization and portability of the system. It will be effective to integrate the micro-size energy harvest and storage units in the same compact system.⁸⁻¹¹

A promising way is to integrate the solar energy harvest and storage parts into one unit sharing the electrodes, thus realizing

a self-powered photo-rechargeable energy unit.¹²⁻²³ A typical solar battery consists of a photo-electrode, a charge-storage electrode, and a counter electrode. In the charging mode, the photo-electrode and charge-storage electrode are connected. After the charging, the charge-storage electrode and counter electrode are connected to drive working devices. In some previous work, only two electrodes can also realize the same functions, where an electrode works as both the photo-electrode and the charge-storage electrode.²⁴⁻²⁷ Such a system may make the photo-charge and discharge process easy and convenient. To date, most of the reports focused on the sandwich-type photo-rechargeable system, and there are rare studies on the fabrication and demonstration of the in-plane-type photo-rechargeable micro-supercapacitors, possibly due to the difficulties on the design and fabrication. An in-plane-type film cell has the advantages of thinner thickness, flexibility, simple production, and the cost in the application to wearable flexible electronic devices. Therefore, it is challenging, but important to explore the suitable materials and methods for the fabrication of such-type photo-rechargeable MSCs.

Carbon materials including carbon nanotubes, graphene, and graphite nanomaterials are recognized as good candidates for flexible high-performance MSCs because of the adaptive properties such as high conductivity, high specific surface area, electrochemical stability, and high mechanical tolerance.²⁸⁻³¹ Many methods such as lithographic and printing techniques have been developed for the fabrication and patterning of the micro-supercapacitors.^{2,32-39} Compared with these methods, laser direct writing, which is a non-contact, efficient, single-step fabrication technique without requirements of masks, post-processing, and complex clean room, is a useful patterning

^aInstitute of Multidisciplinary Research for Advanced Materials, Tohoku University, 2-1-1 Katahira, Aoba-ku, Sendai 980-8577, Japan. E-mail: watanabe@tagen.tohoku.ac.jp

^bInstitute of Materials, China Academy of Engineering Physics, Jiangyou 621908, Sichuan, PR China

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c6ra25136f

technique,^{40–44} and can be easily integrated with current electronic product lines for commercial use. Laser induced carbonization from polymer films was first demonstrated for the fabrication of high-performance carbon MSCs by using high-power CO₂ infrared pulse lasers and 522 nm pulse lasers.^{45–49} Recently, we demonstrated the fabrication of high-performance all-solid-state flexible carbon MSCs by laser direct writing on polyimide (PI) films in air using a relatively low-power compact continuous-wave blue-violet semiconductor laser with a wavelength of 405 nm, the power of which can almost be totally absorbed by the PI film.⁵⁰ We further conducted the similar laser direct writing process on PI film with only changing the air to inert atmosphere with argon (Ar) gas, resulting in highly increased conductivity, specific surface area, and pore size distribution, thus greatly improved energy and power densities.⁵¹ Therefore, laser direct writing is a promising method for the fabrication of high-performance MSCs.

Titanium dioxide (TiO₂) has been widely investigated due to its unique physicochemical properties and many promising applications in energy and environmental fields ranging from solar cells and lithium ion batteries to photocatalysis and sensing.^{52–54} TiO₂ is a very useful model material for the study of photo-rechargeable batteries although it can only absorb UV light. For example, Nomiyama *et al.* reported the sandwich-type photo-rechargeable batteries using TiO₂ nanoparticles as the main photo-absorption materials in the composite films.^{55–57} However, there are rare studies on the in-plane-type photo-rechargeable micro-supercapacitors. In this work, we demonstrated the fabrication of an in-plane-type two-electrode carbon/TiO₂ micro-supercapacitor with photo-rechargeable capability by combining laser direct writing on a PI film in Ar with electrophoretic deposition of TiO₂ nanoparticles on one electrode. The all-solid-state carbon/TiO₂ MSC showed the same excellent capacitive performance with the pure carbon MSC without TiO₂ deposition, such as high specific capacitance, excellent cycling stability, long-time stability, and flexibility. At the same time, this carbon/TiO₂ MSC exhibited strong response to UV light and can be charged to more than 100 mV under UV irradiation. After cycling 10 times, it can still be charged to more than 60 mV without capacitive performance degradation.

2. Experimental

2.1 Preparation of photo-rechargeable micro-supercapacitors

The preparation of carbon electrodes on flexible polyimide (PI) films was conducted using the same method as previously reported.⁵¹ The commercial PI film with a thickness of 125 μm (Kapton 500H) received from Du Pont-Toray Co. Ltd. was first cleaned with water and ethanol before laser irradiation. The laser direct writing process was then carried out in an inert environment of Ar using the same experimental setup as reported in our previous work. A sample box with a quartz window for laser irradiation and a gas inlet and outlet was employed to provide an inert environment by means of an Ar gas flow. A schematic illustration of the laser writing system is shown in Fig. S1.† The laser power was fixed at a typical value of 157 mW

in this work. After laser writing, the samples were rinsed with acetone and water to remove surface dust which was produced as by-products during the laser-induced carbonization. Plasma treatment of the samples was conducted by air-plasma etching for 100 s (JEOL, JFC-1500).

For the preparation of carbon/TiO₂ composite electrodes, laser-irradiated parts of the PI film were first connected to conductive copper tape using Ag paste in order to fabricate into a micro-cell for electrophoretic deposition of TiO₂ nanoparticles. PI tape was employed to define the effective interdigitated area of the electrodes and avoid direct contact between the electrolyte and the electrode pad areas. After air-plasma etching for 100 s (JEOL, JFC-1500) to make the surface hydrophilic, 300 μL of the 6 nm TiO₂ dispersion (1 wt% of an acidic anatase sol TKS201, average diameter 6 nm, TAYCA) was dropped onto the active area. The electrophoretic deposition process was conducted in a two-electrode electrochemical system under a voltage of 3 V for 3 h. After deposition, the active area was cleaned with deionized (DI) water and dried in air. PVA/H₂SO₄ polymer electrolyte was prepared by stirring 1 g of poly(vinyl alcohol) (PVA, *M_w* = 89 000–98 000, Sigma-Aldrich) in 10 mL of DI water and 1 mL of sulfuric acid (98%, Sigma-Aldrich) for 4 hours at 95 °C. The micro-supercapacitors with photo-rechargeable capability were fabricated by dropping the PVA–H₂SO₄ electrolyte (150 μL) onto the active area on the PI substrate and removing excess water by placing the substrate overnight in a desiccator connected to a vacuum pump.

2.2 Characterization

The samples were characterized by scanning electron microscopy (SEM, Hitachi S4800, 10 kV), transmission electron microscopy (TEM, JEOL 2010), and X-ray photoelectron spectroscopy (XPS, Perkin Elmer PHI5600). Raman spectra were measured on a micro-Raman spectrometer equipped with an optical microscope (Olympus BX51), a 532 nm DPSS CW laser (MGL-H-532 nm-1W, CNI), a CCD camera (DV401, Andor Technology), and a monochromator (MS257, Oriel Instruments Co.). The samples for TEM measurement were powders scraped from laser written films. Other characterizations were conducted directly on the films. X-ray diffraction pattern of TiO₂ nanoparticles was collected using a diffractometer (Rigaku Rint-Ultima III) with Cu Kα ($\lambda = 1.54187 \text{ \AA}$) radiation.

2.3 Electrochemical measurements of micro-supercapacitors

Cyclic voltammetry (CV) and galvanostatic charge–discharge (CC) measurements were carried out using an electrochemical workstation (HZ-5000, Hokuto Denko Ltd.) and a DC Voltage and Current Source Monitor 6241A (ADCMT Corporation), respectively. Electrochemical impedance spectra measurements were conducted using a Hioki 3522 LCR HiTESTER over a frequency range from 10 mHz to 100 kHz. For the bending experiments, an MSC was fixed on a series of stainless steel tubes with various curvatures before CV curves were measured.



The calculation of the specific capacitances (C_A , mF cm^{-2}) of the MSCs from the CV curves was based on the following equation:⁵¹

$$C_A = \frac{1}{2 \times A \times v \times (V_f - V_i)} \int_{V_i}^{V_f} I(V) dV \quad (1)$$

where A is the area of the active electrodes (cm^2), v is the scan rate (V s^{-1}), V_f and V_i are the vertex potentials of the CV scan, and $I(V)$ is the current as a function of potential (A). $\int_{V_i}^{V_f} I(V) dV$ is the numerically integrated area of the CV curve.

The calculation of the specific capacitances (C_A , mF cm^{-2}) of the MSCs from the CC curves was based on the following equation:⁵¹

$$C_A = \frac{I}{A \times dV/dt} \quad (2)$$

where I is the discharge current (A), A is the projected area of active electrodes (cm^2), and dV/dt is the slope of the galvanostatic discharge curves.

2.4 Photo-rechargeable capability measurements of carbon/TiO₂ micro-supercapacitors

Two-electrode electrochemical system was employed to measure the photo-rechargeable capability of the carbon/TiO₂ MSC with positive electrode and negative electrode connected to carbon/TiO₂ electrode and opposite carbon electrode, respectively. The UV light was provided by a 150 W xenon lamp (L2274, HAMAMATSU) with a UV transmitting filter (UTVAF-50S-33U, SIGMA KOKI Japan), and a PC-controlled shutter unit (F77-6, SURUGA SEIKI) was used to switch the light on and off.

3. Results and discussion

3.1 Morphology and structure

A schematic illustration of the fabrication process for the carbon/TiO₂ composite electrode is shown in Fig. 1. The interdigitated carbon electrodes were first fabricated by laser direct writing on a clean polyimide (PI) film in Ar at a typical laser power of 157 mW. The carbon electrodes were then connected to the external circuits with Cu tape and Ag paste, which are protected by PI tape to avoid the contact between electrodes and electrolyte. Air-plasma etching was employed to modify the

surface wettability of the carbon electrodes from hydrophobic to superhydrophilic. After 300 μL of the TiO₂ dispersion (1 wt%) dropped onto the active area, the electrophoretic deposition process was conducted in a two-electrode electrochemical system under a voltage of 3 V for 3 h. It should be noted that as-received TiO₂ nanoparticles are anatase phase and have a diameter less than ten nanometers, which are also confirmed by the XRD pattern and TEM image (Fig. S2†). After deposition, the electrodes were cleaned with DI water and PVA/H₂SO₄ electrolyte (150 μL) was dropped onto the active area on the PI substrate. After removing excess water by placing the substrate overnight in a desiccator connected to a vacuum pump, the all-solid-state asymmetric MSC with one carbon/TiO₂ electrode and one carbon electrode was prepared successfully.

The carbon layer obtained by laser direct writing in Ar at a typical laser power of 157 mW before electrophoretic deposition of TiO₂ nanoparticles shows the same structures with our previous work, exhibiting a smooth surface with many internal nano- and micro-pores and a thickness of about 12 μm (Fig. S3†). The areal loading of the carbon materials is about 0.8 mg cm^{-2} , which was calculated by weighing the carbon powder scraped from the laser-writing parts. The current curve during the electrophoretic deposition process is shown in Fig. S4.† It should be noted that hydrogen and oxygen gases are evolved from the electrodes during the electrophoretic deposition process because of the relatively high voltage. Fig. 2 shows the SEM images of the sample after electrophoretic deposition of TiO₂ nanoparticles. In Fig. 2a, the top dark branch is the negative electrode, on which there are only few TiO₂ nanoparticles on the surface because of the physical adsorption. Compared with the negative electrode, the positive electrode of the bottom branch became relatively bright in the SEM image. It also shows a smooth surface although there are some cracks due to stress in the carbon structures (Fig. 2b). An enlarged SEM image in Fig. 2c indicates a thick layer of TiO₂ nanoparticles on the surface. The high-resolution SEM image in Fig. 2d clearly shows the TiO₂ nanoparticles on the surface. The thickness of the TiO₂ layer was less than 200 nm from the cross-sectional SEM image although the exact determination of the thickness is difficult because of the rough surface of the porous carbon layer. Considering the average size of TiO₂ nanoparticles from TEM image (*ca.* 6 nm), the areal loading of the TiO₂ layer was estimated to be less than 0.06 mg cm^{-2} , which is ten times lower than that of the carbon layer.

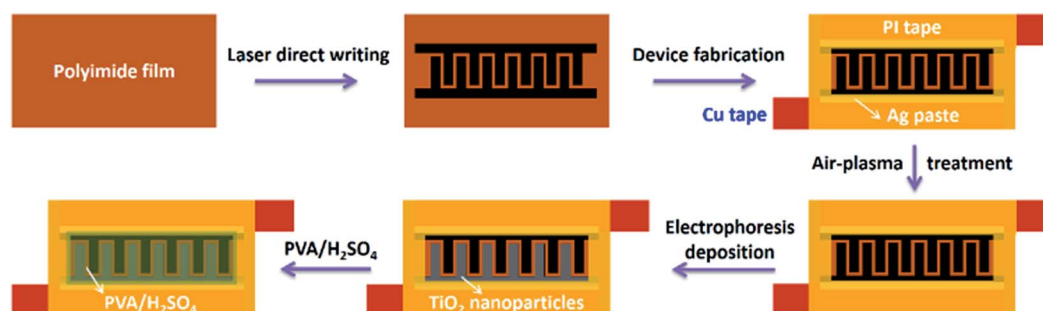


Fig. 1 Schematic illustration for the fabrication of a carbon/TiO₂ micro-supercapacitor with photo-rechargeable capability.



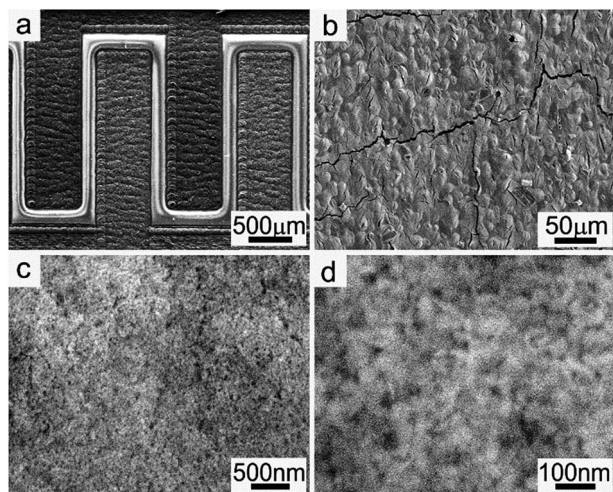


Fig. 2 Large-scale (a and b) and enlarged SEM images (c and d) of the carbon/TiO₂ composite electrode obtained by laser direct writing on PI film in Ar and following electrophoretic deposition of TiO₂ nanoparticles.

The typical Raman spectra of carbon structures and carbon/TiO₂ composites are shown in Fig. 3a. The Raman peaks at 1350 cm⁻¹ and 1580 cm⁻¹ observed in both samples are corresponded to the D band and G band for graphitized or amorphous carbon materials, but the intensity for carbon/TiO₂ composite structures was suppressed largely due to the cover of TiO₂ nanoparticles. The Raman peaks at 161, 407, 519, and 638 cm⁻¹ in carbon/TiO₂ composite electrodes can be assigned as the E_g, B_{1g}, B_{1g} or A_{1g}, E_g modes of the anatase phase, respectively,⁵⁸ thus confirming the deposition of TiO₂ nanoparticles on the carbon surface. XPS spectra in Fig. 3b indicate the carbon peaks were greatly reduced and the Ti and O peaks become dominated, further demonstrating the formation of TiO₂-on-carbon composite structures. Therefore, the carbon/TiO₂ electrode can be facily prepared by combining laser direct writing technique with electrophoretic deposition of TiO₂ nanoparticles.

3.2 Capacitive performance of the carbon/TiO₂ micro-supercapacitor

After electrophoretic deposition of TiO₂ nanoparticles, PVA/H₂SO₄ electrolyte was dropped onto the active area and dried, forming an asymmetric MSC with a carbon/TiO₂ electrode and pure carbon electrode (Fig. S5†). The electrochemical performance of the carbon/TiO₂ MSC was first evaluated by cyclic voltammetry (CV) and galvanostatic charge-discharge (CC) measurements. Fig. 4a shows the CV curves at low scan rates for the typical carbon/TiO₂ MSC prepared by laser direct writing in Ar at 157 mW with air-plasma treatment for 100 s and electrophoretic deposition of TiO₂ nanoparticles at 3 V for 3 h, which exhibit stable quasi-rectangular shapes, indicating electrochemical double-layer (EDL) capacitive properties. The CV curves at relatively high scan rates also suggest apparent EDL capacitive properties (Fig. 4b). Fig. 4c plots the specific areal capacitances (C_A) calculated from the CV curves as a function of



Fig. 3 Raman (a) and XPS spectra (b) of the structures before and after electrophoretic deposition of TiO₂ nanoparticles.

scan rate. The C_A at a typical scan rate of 10 mV s⁻¹ is 15.5 mF cm⁻², which is comparable with the 16 mF cm⁻² for the pure carbon electrodes. The carbon/TiO₂ MSC shows a little lower capacitance than pure carbon electrodes at low scan rates, but higher capacitances at high scan rates. The CC curves of the carbon/TiO₂ electrode are shown in Fig. 4d, all of which exhibit capacitive triangular shapes. The corresponding C_A values calculated from the CC curves as a function of current density is shown in Fig. 4e. The typical capacitance at a current density of 0.05 mA cm⁻² is 27.3 mF cm⁻², which is a little lower than the 28.8 mA cm⁻² for pure carbon MSC with air-plasma treatment. The carbon/TiO₂ MSC shows a little lower specific capacitance with the pure carbon MSC at low current densities, but higher capacitances at high current densities (Fig. 4e). It should be noted that the typical capacitances calculated from either CV curve or CC curve are higher than those for most of the MSCs based on carbon materials, even for carbon nanotubes and graphene. The electrochemical impedance spectra (EIS) of both carbon/TiO₂ MSC and pure carbon MSC are shown in Fig. 4f. The relatively vertical slope of the Nyquist plot for carbon/TiO₂ MSC in the low-frequency region suggests better capacitive behaviour than pure carbon MSC. Although the intercept on the real axis for carbon/TiO₂ MSC is 123 Ω, higher than the 52 Ω for pure carbon MSC, indicating that carbon/TiO₂ MSC has a larger



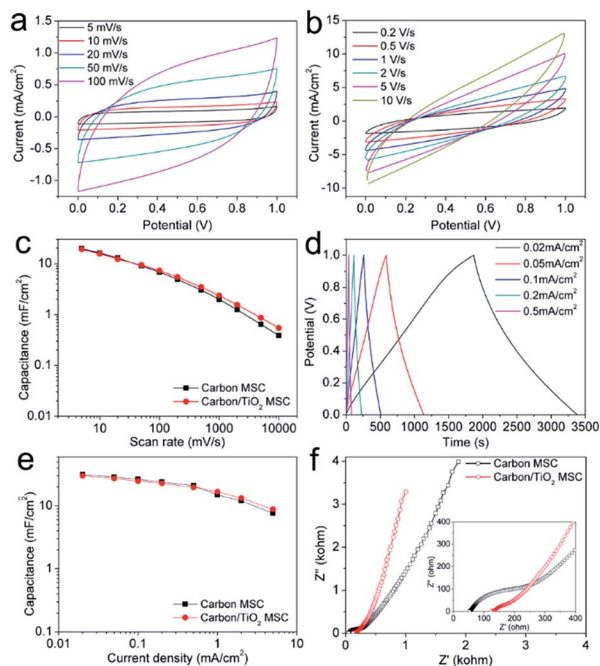


Fig. 4 Electrochemical properties of the carbon/TiO₂ composite electrode. CV curves at low scan rate (a) and high scan rate (b), specific capacitance calculated from CV data as a function of scan rate (c), CC curves (d), specific capacitance calculated from CC data as a function of current density (e), and Nyquist plots of MSCs before and after TiO₂ deposition (f). The inset in (f) shows the high-frequency region of the Nyquist plots.

equivalent series resistance, a shorter Warburg region and smaller diameter semicircle in the high- and mid-frequency regions for carbon/TiO₂ MSC suggest a lower charge-transfer resistance and a more efficient ion diffusion process than those for pure carbon MSC. Therefore, the carbon/TiO₂ MSC exhibits a comparable capacitive performance with pure carbon MSC.

The cycling stability, long-time stability, and mechanical stability were studied, which are very important for practical use. The cycling stability of carbon/TiO₂ MSC was measured over 10 000 CV cycles at a scan rate of 50 mV s⁻¹, showing no capacitance degradation (Fig. 5a). The excellent cycling stability can be ascribed to the high stability of the carbon/TiO₂ electrode and the interface between the electrode and the electrolyte. The small change of the CV shapes after the cycling measurement may imply that the interface between carbon/TiO₂ electrodes and the electrolyte became more stable and capacitive with cycling scan (Fig. 5b). Surprisingly, the carbon/TiO₂ MSC shows an increase specific capacitance and a more capacitive behaviour even after 3 months kept in air (Fig. 5c), which is much better than the MSC based on multi-walled carbon nanotubes with PVA/H₃PO₄ electrolyte.⁵⁹ This result suggests that the carbon/TiO₂ layered hybrid structures are very stable under the acidic condition and the MSC becomes more capacitive due to the osmosis of electrolyte through the pores in the carbon electrode. The mechanical stability of a typical carbon/TiO₂ MSC was evaluated by bending the MSC tightly on

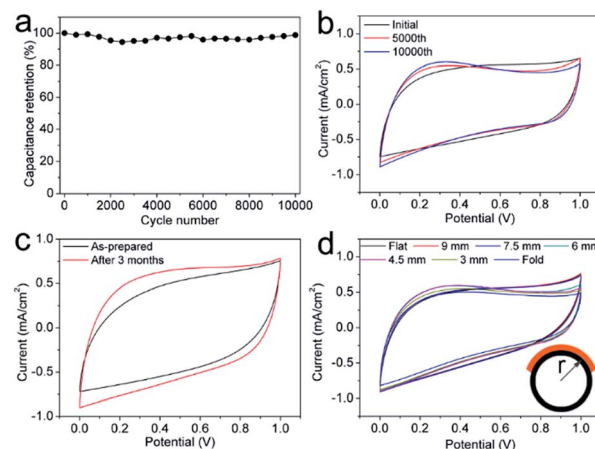


Fig. 5 (a) Cycling stability at a scan rate of 50 mV s⁻¹, (b) CV curves of the initial, 5000th, and 10 000th cycles, (c) CV curves of the as-prepared C/TiO₂ MSC and the MSC after 3 months (scan rate 50 mV s⁻¹), and (d) CV curves of MSCs bent with different curvature radii (scan rate 50 mV s⁻¹). Inset in (d) is a schematic picture of a single MSC bent at different curvature radii.

a series of stainless steel tubes with different diameters. The CV curves in Fig. 5d indicate similar capacitive shapes with no apparent capacitance degradation, even for the MSC bent to a curvature radius of 3 mm and folded 180 degrees, suggesting an excellent flexibility and mechanical stability. Therefore, the carbon/TiO₂ MSC shows excellent cycling stability, long-time stability, and flexibility.

3.3 Photo-rechargeable capability of the carbon/TiO₂ micro-supercapacitor

The evaluation for the photo-rechargeable capability of the carbon/TiO₂ MSC was carried out using an electrochemical workstation in dark and under UV light irradiation. The UV light was provided by a Xe lamp with a UV transmitting filter (UTVAF-50S-33U, Sigmakoki, Japan), and a shutter was employed to control the light on and off. Fig. 6a shows the current density of the typical carbon/TiO₂ electrode in dark and under UV light irradiation. The MSC in dark shows a current of about 0.5 $\mu\text{A cm}^{-2}$ at first and it gradually decreased, which can be attributed to the slow balance process of the potential difference between the carbon/TiO₂ electrode and the opposite carbon electrode. Under UV light irradiation, the MSC exhibits a stable current of about 1.6 $\mu\text{A cm}^{-2}$, three times of the 0.5 $\mu\text{A cm}^{-2}$ in dark, suggesting a photocurrent induced by the UV light irradiation. Fig. 6b shows the current changes of the carbon/TiO₂ MSC when the UV light was switched on or off, which exhibits a fast and stable response to UV light. The potential change between two electrodes was also monitored when the UV light was on or off. As shown in Fig. 6c, the carbon/TiO₂ MSC in dark shows a low potential of about 13 mV due to the interfacial potential difference between two electrodes. On the other hand, when the UV light turned on, the potential was increased to more than 100 mV within 10 min. If the light turned off, the potential was dropped gradually due to the self-





Fig. 6 (a) Current density of the carbon/TiO₂ MSC in dark and under UV light irradiation, (b) current change curve of the carbon/TiO₂ MSC by switching the UV light on and off, (c) potential change of the carbon/TiO₂ MSC in dark (black line) and with UV light on and off (red), (d) photo-charge under UV light irradiation and discharge process at a current density of 1 $\mu\text{A cm}^{-2}$, (e) the potential change at different photo-charge and discharge process, and (f) CV curves of initial cycle and MSC after 10 photo-charge and discharge cycles (scan rate 5 mV s^{-1}).

discharge, indicating that the potential also shows fast and stable response to the UV light. Fig. 6d shows the potential change curve of the carbon/TiO₂ MSC during the charge process under UV light irradiation without external applied current for 500 s and discharge process at a current density of 1 $\mu\text{A cm}^{-2}$. Compared with the control potential change curve without UV light irradiation, the carbon/TiO₂ MSC shows a much higher potential, suggesting the photo-chargable capability of the carbon/TiO₂ MSC. The cycling properties were also studied. As shown in Fig. 6e, the carbon/TiO₂ MSC can still be charged to more than 60 mV under UV light after 10 cycles, indicating a photo-rechargeable capability. The voltage decay in the initial several cycles may be caused by the TiO₂/electrolyte interface change due to ion diffusion after UV light irradiation, and the interface gradually became stable after several photo-charging cycles. It should be noted that the capacitive performance shows no degradation after 10 cycles, suggesting the stability of the carbon/TiO₂ MSC structures under UV light (Fig. 6f).

A tentative photo-charging mechanism was proposed as shown in Fig. 7. When the UV light is switched on, the TiO₂ nanoparticles absorb the UV light and the valence electrons are excited to the conduction band. Then the excited electrons are transferred to the underneath carbon electrode and then to the opposite carbon electrode, finally stored in the interface between carbon structures and electrolyte in the manner of electrochemical double-layer; while the holes may be stored at the interface between TiO₂ nanoparticles and the electrolyte in



Fig. 7 A schematic illustration of the tentative photo-charge and discharge mechanism.

the same manner or transferred to the electrolyte. In the discharge state, the electrons stored in the carbon electrode will move back to the carbon/TiO₂ electrode to drive the external circuit. Therefore, we demonstrated a new in-plane-type carbon/TiO₂ micro-supercapacitor with photo-rechargeable capability by combining laser direct writing technique for patterning carbon electrodes with electrophoretic deposition of TiO₂ nanoparticles, although the performance is capable of improvement. The relatively low voltage charged by UV light irradiation may be due to the thin thickness of the TiO₂ nanoparticle film limited by the electrophoretic deposition. On one hand, the low areal amount of TiO₂ nanoparticles leads to insufficient absorption of UV light, thus the relatively low voltage. In addition, the low areal amount of TiO₂ nanoparticles cannot provide enough surface area to store the photo-generated holes. Therefore, with increase of the thickness of the TiO₂ layer by other methods such as drop casting followed by laser patterning or electrochemical deposition, more UV light may be absorbed and stored, thus the MSC can be charged to a higher potential. Besides, other visible-light-absorption materials or combination systems which can absorb visible light, may store more photo energy effectively. Therefore, it can be expected that with careful design for structures and materials the high-performance photo-rechargeable micro-supercapacitors can be patterned by the laser direct writing technique followed by the selective deposition methods such as electrochemical deposition or layer-by-layer laser direct writing method.

4. Conclusion

We demonstrated the fabrication of high-performance carbon/TiO₂ micro-supercapacitors with photo-rechargeable capability by combining laser direct writing on polyimide film in Ar with electrophoretic deposition of TiO₂ nanoparticles. The carbon/TiO₂ MSC exhibits a comparable capacitive performance as pure carbon MSC, showing a typical specific capacitance of 27.3 mF cm^{-2} at a current density of 0.05 mA cm^{-2} , excellent cycling stability, long-time stability, as well as mechanical stability. The carbon/TiO₂ MSC can be charged to above 100 mV under UV



light irradiation, and still keep 60 mV after 10 photo-charging cycles, demonstrating the photo-rechargeable capability. Although the photo-chargeable performance of this new type device was capable of improvement, it is expected that this work can provide some thoughts for the fabrication of high-performance photo-rechargeable micro-supercapacitors, even totally self-powered in-plane-type flexible systems.

Acknowledgements

This work was supported by a Grant-in-Aid for Scientific Research on Innovative Areas "New Polymeric Materials Based on Element Blocks (No. 2401)" (JSPS KAKENHI Grant Number JP24102004) and JSPS KAKENHI Grant Number JP15H04132, National Natural Science Foundation of China (No. 21603201), China Academy of Engineering Physics (item no. TP201302-3), and the Fundamental Application Research of the Department of Science and Technology of Sichuan Province (Grant no. 2014JY0137). The authors thank Mr Eiji Aoyagi of the Electron Microscopy Center in Tohoku University for help with the SEM and TEM measurements. The authors also thank Ms Sayaka Ogawa for help with the XPS measurement.

References

- 1 B. C. Kim, J.-Y. Hong, G. G. Wallace and H. S. Park, *Adv. Energy Mater.*, 2015, **5**, 1500959.
- 2 Z.-S. Wu, X. Feng and H.-M. Cheng, *Natl. Sci. Rev.*, 2014, **1**, 277–292.
- 3 M. Beidaghi and Y. Gogotsi, *Energy Environ. Sci.*, 2014, **7**, 867–884.
- 4 H. Xia, C. Hong, B. Li, B. Zhao, Z. Lin, M. Zheng, S. V. Savilov and S. M. Aldoshin, *Adv. Funct. Mater.*, 2015, **25**, 627–635.
- 5 J. Liu, M. Zheng, X. Shi, H. Zeng and H. Xia, *Adv. Funct. Mater.*, 2016, **26**, 919–930.
- 6 H. Gao and K. Lian, *RSC Adv.*, 2014, **4**, 33091–33113.
- 7 A. Tyagi, K. M. Tripathi and R. K. Gupta, *J. Mater. Chem. A*, 2015, **3**, 22507–22541.
- 8 G. Wee, T. Salim, Y. M. Lam, S. G. Mhaisalkar and M. Srinivasan, *Energy Environ. Sci.*, 2011, **4**, 413–416.
- 9 T. Chen, L. Qiu, Z. Yang, Z. Cai, J. Ren, H. Li, H. Lin, X. Sun and H. Peng, *Angew. Chem., Int. Ed.*, 2012, **51**, 11977–11980.
- 10 C.-T. Chien, P. Hiralal, D.-Y. Wang, I. S. Huang, C.-C. Chen, C.-W. Chen and G. A. J. Amaratunga, *Small*, 2015, **11**, 2929–2937.
- 11 J. Xu, Y. Chen and L. Dai, *Nat. Commun.*, 2015, **6**, 8103.
- 12 T. Chen, Z. Yang and H. Peng, *ChemPhysChem*, 2013, **14**, 1777–1782.
- 13 Z. Yang, L. Li, Y. Luo, R. He, L. Qiu, H. Lin and H. Peng, *J. Mater. Chem. A*, 2013, **1**, 954–958.
- 14 X. Chen, H. Sun, Z. Yang, G. Guan, Z. Zhang, L. Qiu and H. Peng, *J. Mater. Chem. A*, 2014, **2**, 1897–1902.
- 15 X. Huang, X. Zhang and H. Jiang, *J. Power Sources*, 2014, **248**, 434–438.
- 16 A. S. Westover, K. Share, R. Carter, A. P. Cohn, L. Oakes and C. L. Pint, *Appl. Phys. Lett.*, 2014, **104**, 213905.
- 17 J. Xu, H. Wu, L. Lu, S.-F. Leung, D. Chen, X. Chen, Z. Fan, G. Shen and D. Li, *Adv. Funct. Mater.*, 2014, **24**, 1840–1846.
- 18 Z. Yang, J. Deng, H. Sun, J. Ren, S. Pan and H. Peng, *Adv. Mater.*, 2014, **26**, 7038–7042.
- 19 Z. Zhang, X. Chen, P. Chen, G. Guan, L. Qiu, H. Lin, Z. Yang, W. Bai, Y. Luo and H. Peng, *Adv. Mater.*, 2014, **26**, 466–470.
- 20 N. Li, Y. Wang, D. Tang and H. Zhou, *Angew. Chem., Int. Ed.*, 2015, **54**, 9271–9274.
- 21 Q. Li, N. Li, M. Ishida and H. Zhou, *J. Mater. Chem. A*, 2015, **3**, 20903–20907.
- 22 D. Schmidt, M. D. Hager and U. S. Schubert, *Adv. Energy Mater.*, 2016, **6**, 1500369.
- 23 M. Yu, W. D. McCulloch, Z. Huang, B. B. Trang, J. Lu, K. Amine and Y. Wu, *J. Mater. Chem. A*, 2016, **4**, 2766–2782.
- 24 X. Zhang, X. Huang, C. Li and H. Jiang, *Adv. Mater.*, 2013, **25**, 4093–4096.
- 25 X. Zou, N. Maesako, T. Nomiyama, Y. Horie and T. Miyazaki, *Sol. Energy Mater. Sol. Cells*, 2000, **62**, 133–142.
- 26 T. Miyasaka and T. N. Murakami, *Appl. Phys. Lett.*, 2004, **85**, 3932–3934.
- 27 Q. Wang, H. Chen, E. McFarland and L. Wang, *Adv. Energy Mater.*, 2015, **5**, 1501418.
- 28 P. Simon and Y. Gogotsi, *Nat. Mater.*, 2008, **7**, 845–854.
- 29 Y. He, W. Chen, C. Gao, J. Zhou, X. Li and E. Xie, *Nanoscale*, 2013, **5**, 8799–8820.
- 30 Z. Yu, L. Tetard, L. Zhai and J. Thomas, *Energy Environ. Sci.*, 2015, **8**, 702–730.
- 31 D. Yu, K. Goh, H. Wang, L. Wei, W. Jiang, Q. Zhang, L. Dai and Y. Chen, *Nat. Nanotechnol.*, 2014, **9**, 555–562.
- 32 J. Chmiola, C. Largeot, P.-L. Taberna, P. Simon and Y. Gogotsi, *Science*, 2010, **328**, 480–483.
- 33 J. Chmiola, G. Yushin, Y. Gogotsi, C. Portet, P. Simon and P. L. Taberna, *Science*, 2006, **313**, 1760–1763.
- 34 K. U. Laszcyk, K. Kobashi, S. Sakurai, A. Sekiguchi, D. N. Futaba, T. Yamada and K. Hata, *Adv. Energy Mater.*, 2015, **5**, 1500741.
- 35 Z. Liu, Z.-S. Wu, S. Yang, R. Dong, X. Feng and K. Müllen, *Adv. Mater.*, 2016, **28**, 2217–2222.
- 36 Z. Niu, L. Zhang, L. Liu, B. Zhu, H. Dong and X. Chen, *Adv. Mater.*, 2013, **25**, 4035–4042.
- 37 D. Pech, M. Brunet, H. Durou, P. Huang, V. Mochalin, Y. Gogotsi, P.-L. Taberna and P. Simon, *Nat. Nanotechnol.*, 2010, **5**, 651–654.
- 38 Z. S. Wu, K. Parvez, X. Feng and K. Müllen, *Nat. Commun.*, 2013, **4**, 2487.
- 39 Z.-S. Wu, K. Parvez, A. Winter, H. Vieker, X. Liu, S. Han, A. Turchanin, X. Feng and K. Müllen, *Adv. Mater.*, 2014, **26**, 4552–4558.
- 40 G. Qin and A. Watanabe, *J. Nanopart. Res.*, 2014, **16**, 2684.
- 41 G. Qin, L. Fan and A. Watanabe, *J. Mater. Process. Technol.*, 2016, **227**, 16–23.
- 42 Q. Gang, W. Akira, T. Hiroki and Y. Tetsu, *Jpn. J. Appl. Phys.*, 2014, **53**, 096501.
- 43 M. Aminuzzaman, A. Watanabe and T. Miyashita, *J. Electron. Mater.*, 2015, **44**, 4811–4818.
- 44 M. Aminuzzaman, A. Watanabe and T. Miyashita, *J. Nanopart. Res.*, 2010, **12**, 931–938.



- 45 Z. Peng, R. Ye, J. A. Mann, D. Zakhidov, Y. Li, P. R. Smalley, J. Lin and J. M. Tour, *ACS Nano*, 2015, **9**, 5868–5875.
- 46 Z. Peng, J. Lin, R. Ye, E. L. G. Samuel and J. M. Tour, *ACS Appl. Mater. Interfaces*, 2015, **7**, 3414–3419.
- 47 J. Lin, Z. Peng, Y. Liu, F. Ruiz-Zepeda, R. Ye, E. L. G. Samuel, M. J. Yacaman, B. I. Yakobson and J. M. Tour, *Nat. Commun.*, 2014, **5**, 5714.
- 48 L. Li, J. Zhang, Z. Peng, Y. Li, C. Gao, Y. Ji, R. Ye, N. D. Kim, Q. Zhong, Y. Yang, H. Fei, G. Ruan and J. M. Tour, *Adv. Mater.*, 2016, **28**, 838–845.
- 49 J. B. In, B. Hsia, J.-H. Yoo, S. Hyun, C. Carraro, R. Maboudian and C. P. Grigoropoulos, *Carbon*, 2015, **83**, 144–151.
- 50 J. Cai, C. Lv and A. Watanabe, *J. Mater. Chem. A*, 2016, **4**, 1671–1679.
- 51 J. Cai, C. Lv and A. Watanabe, *Nano Energy*, 2016, **30**, 790–800.
- 52 J. Ye, W. Liu, J. Cai, S. Chen, X. Zhao, H. Zhou and L. Qi, *J. Am. Chem. Soc.*, 2010, **133**, 933–940.
- 53 X. Zhao, W. Jin, J. Cai, J. Ye, Z. Li, Y. Ma, J. Xie and L. Qi, *Adv. Funct. Mater.*, 2011, **21**, 3554–3563.
- 54 J. Cai, J. Ye, S. Chen, X. Zhao, D. Zhang, S. Chen, Y. Ma, S. Jin and L. Qi, *Energy Environ. Sci.*, 2012, **5**, 7575–7581.
- 55 X. Zou, T. Nagao, O. Miyamoto, T. Nomiyama, Y. Horie and T. Miyazaki, *Jpn. J. Appl. Phys.*, 2004, **43**, 7707.
- 56 H. Usui, O. Miyamoto, T. Nomiyama, Y. Horie and T. Miyazaki, *Sol. Energy Mater. Sol. Cells*, 2005, **86**, 123–134.
- 57 N. Teruaki, S. Kenichi, S. Kenta and H. Yuji, *Jpn. J. Appl. Phys.*, 2015, **54**, 071101.
- 58 P. Song, X. Zhang, M. Sun, X. Cui and Y. Lin, *Nanoscale*, 2012, **4**, 1800–1804.
- 59 D. Kim, G. Lee, D. Kim and J. S. Ha, *ACS Appl. Mater. Interfaces*, 2015, **7**, 4608–4615.

